

Geometric Patterns and Microstructures in the study of Material Defects and Composites

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Department of Mathematics



Presentation Plan

① Geometric Patterns of Dislocations

② Microstructures in Composites

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- ▶ Dislocations

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- ▶ Dislocations
- ▶ Semi-coherent interfaces (Chapter 3)

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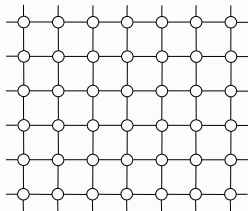
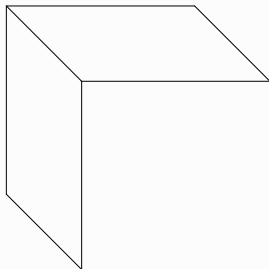
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② Microgeometries in Composites

- ▶ Critical lower integrability
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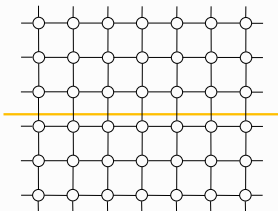
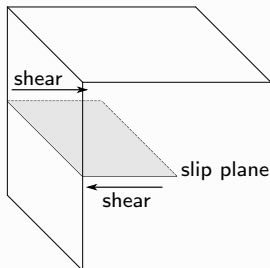
Edge dislocations

Dislocations: topological defects in the otherwise periodic structure of a crystal.



Edge dislocations

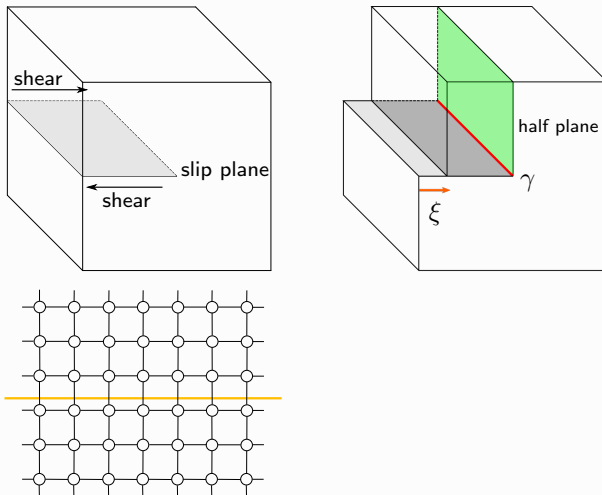
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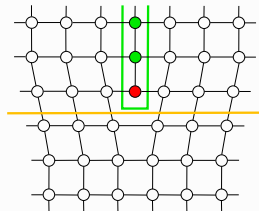
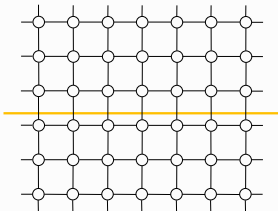
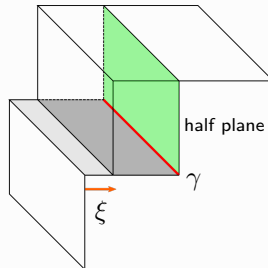
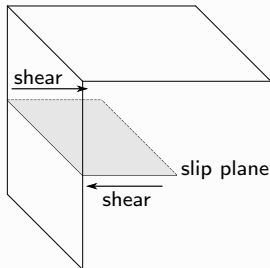
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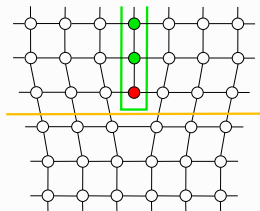
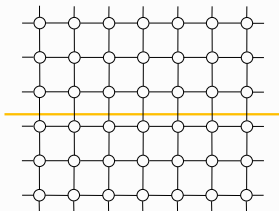
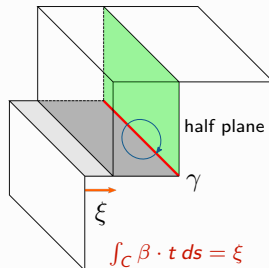
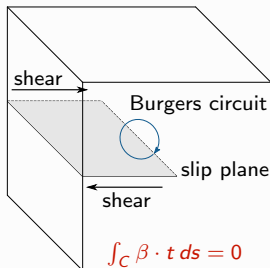
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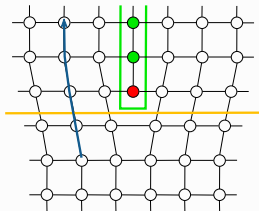
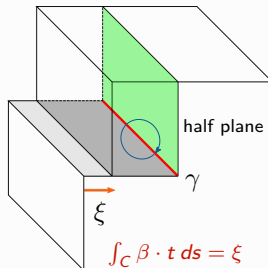
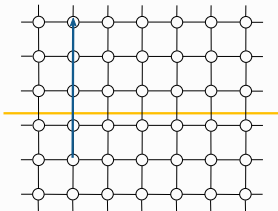
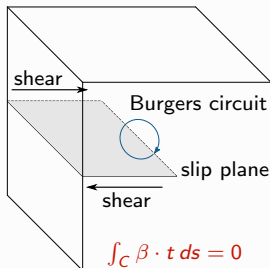
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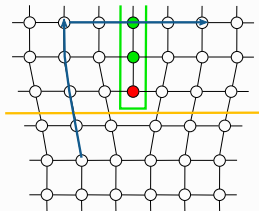
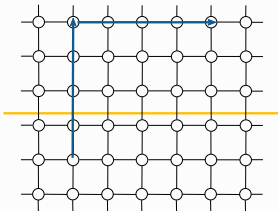
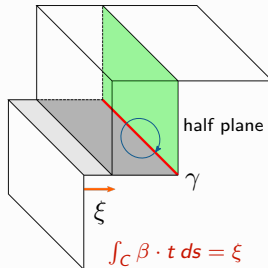
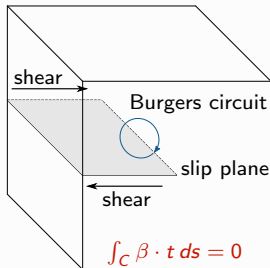
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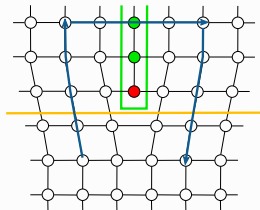
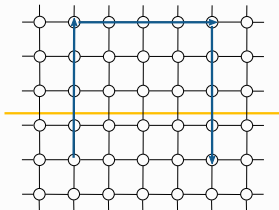
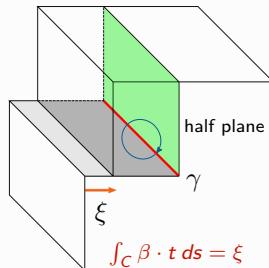
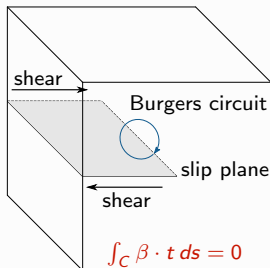
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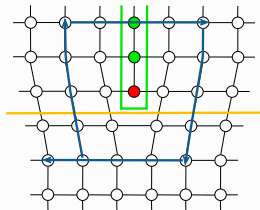
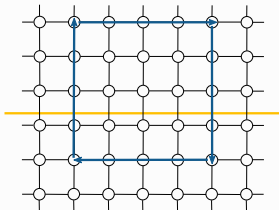
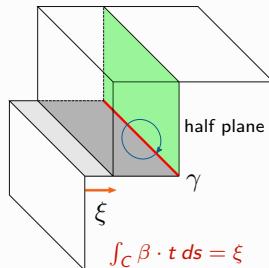
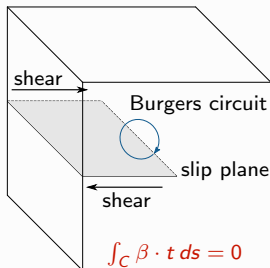
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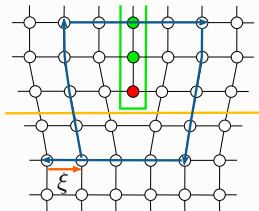
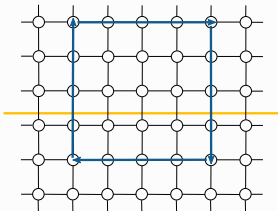
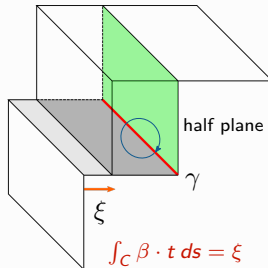
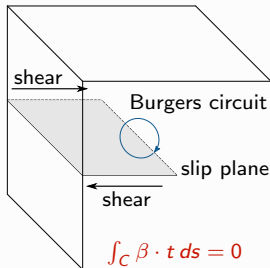
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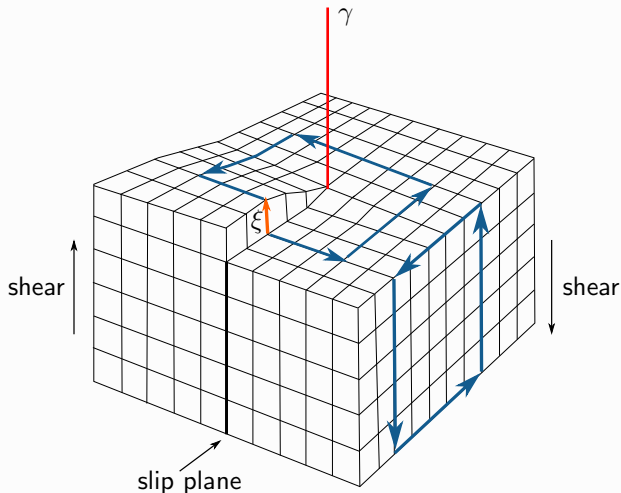


Screw dislocations

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Mixed type dislocations

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Mixed type dislocations

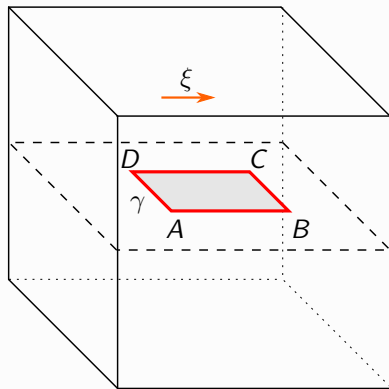
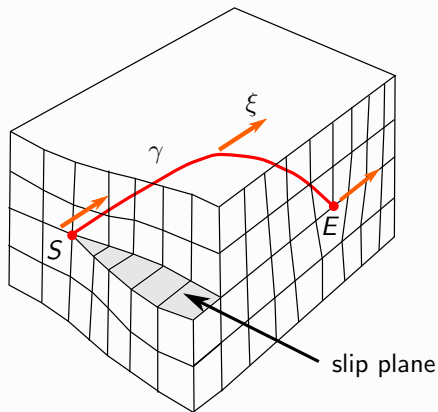
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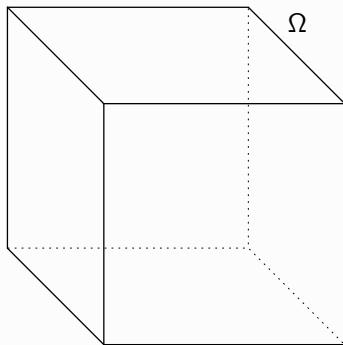
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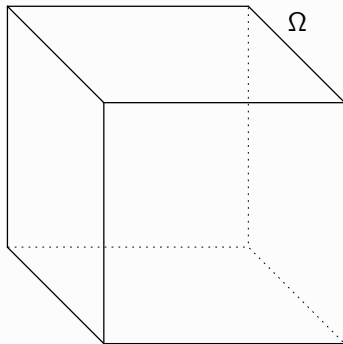
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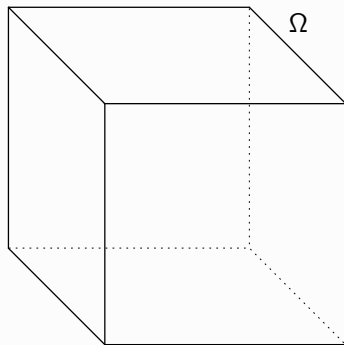


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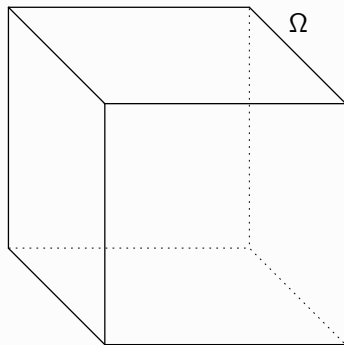
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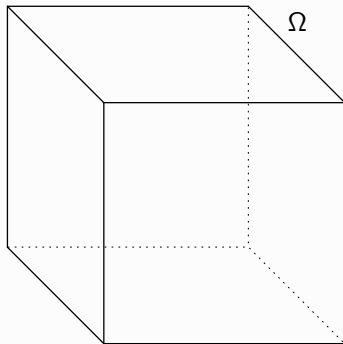
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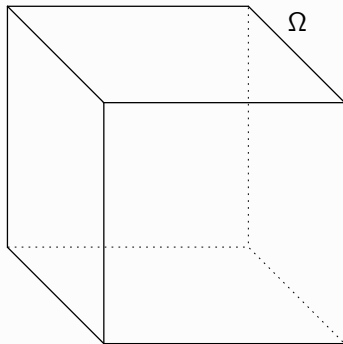
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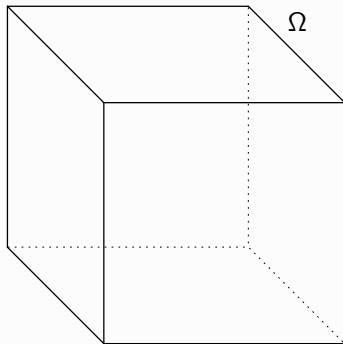
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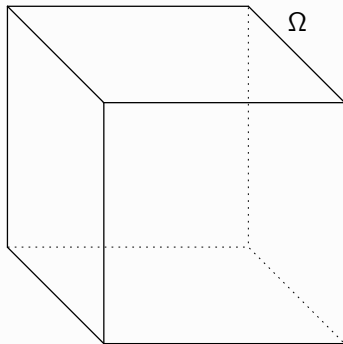
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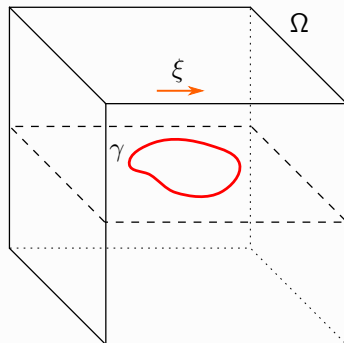
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Semi-discrete model for dislocations

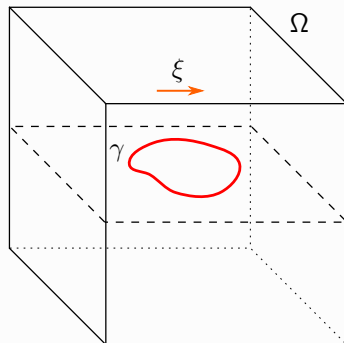
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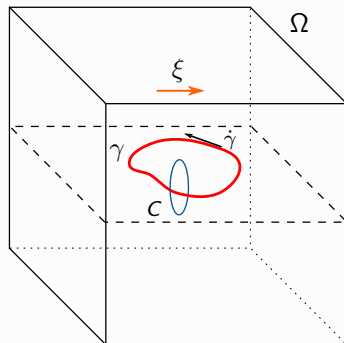
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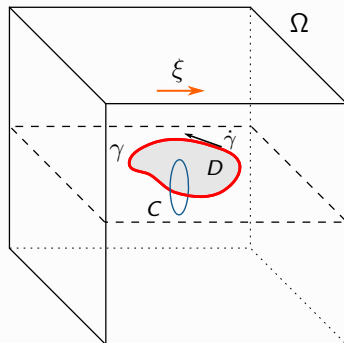
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Geometric interpretation: if D encloses γ , there exists a deformation $v \in SBV(\Omega; \mathbb{R}^3)$ s.t.

$$Dv = \nabla v \, dx + \xi \otimes n \mathcal{H}^2 \llcorner D, \quad \beta = \nabla v.$$

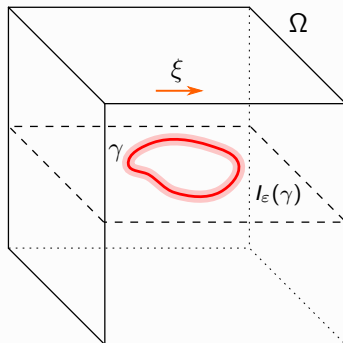
v has constant jump ξ across the slip region D .



Strains are not L^2

Let β generate (γ, ξ) . Consider $\varepsilon > 0$ and

$$I_\varepsilon(\gamma) := \{x \in \mathbb{R}^3 : \text{dist}(x, \gamma) < \varepsilon\}.$$



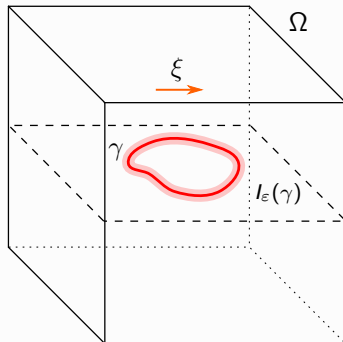
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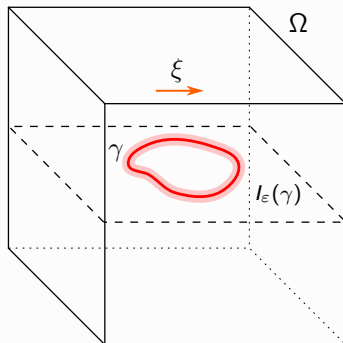
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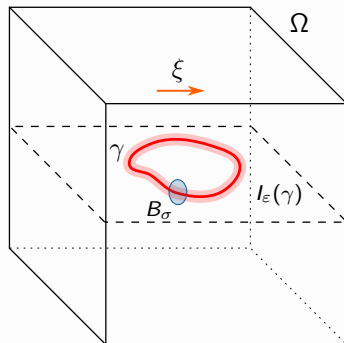
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$$\int_{I_\sigma \setminus I_\varepsilon} |\beta|^2 = L \int_\varepsilon^\sigma \int_{\partial B_\rho(\gamma(s))} |\beta|^2 d\mathcal{H}^1 d\rho$$



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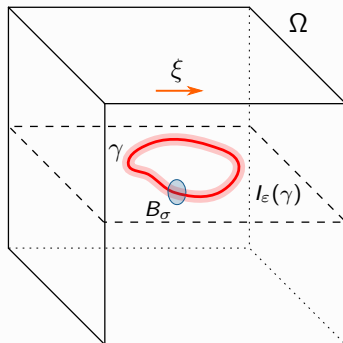
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$$(\text{Jensen}) \geq L \int_\varepsilon^\sigma \frac{1}{2\pi\rho} \left| \int_{\partial B_\rho(\gamma(s))} \beta \cdot t d\mathcal{H}^1 \right|^2 d\rho$$



Strains are not L^2

Let β generate (γ, ξ) . Consider $\varepsilon > 0$ and

$$I_\varepsilon(\gamma) := \{x \in \mathbb{R}^3 : \text{dist}(x, \gamma) < \varepsilon\}.$$

Then we have

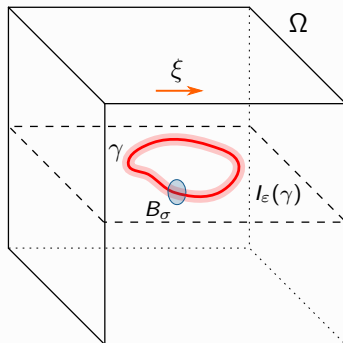
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$$= L \frac{|\xi|^2}{2\pi} \log \frac{\sigma}{\varepsilon} \rightarrow \infty \text{ as } \varepsilon \rightarrow 0$$



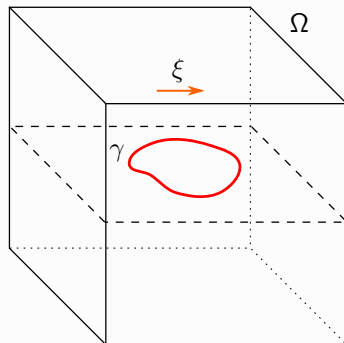
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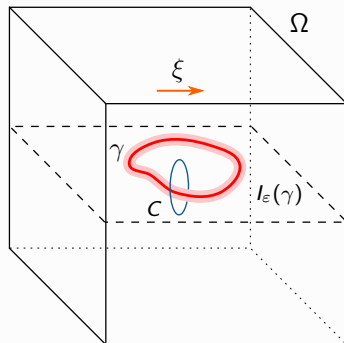
$$W(F) \sim \text{dist}(F, SO(3))^2.$$

Let $\varepsilon > 0$ (\propto atomic distance) and consider

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$$\text{Curl } \beta \llcorner \Omega_\varepsilon(\gamma) = 0, \quad \int_C \beta \cdot t \, d\mathcal{H}^1 = \xi.$$



Presentation Plan

① Geometric Patterns of Dislocations

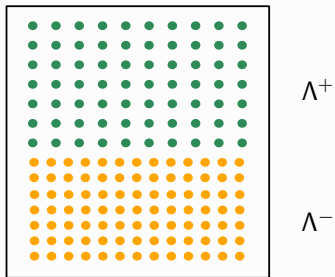
- ▶ Dislocations
- ▶ Semi-coherent interfaces
- ▶ Linearised polycrystals

② Microgeometries in Composites

- ▶ Critical lower integrability
- ▶ Convex integration
- ▶ Proof of our main result

Semi-coherent interfaces

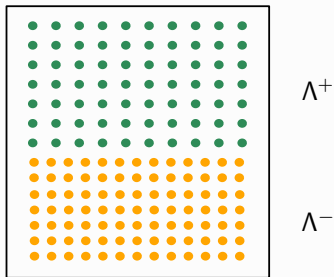
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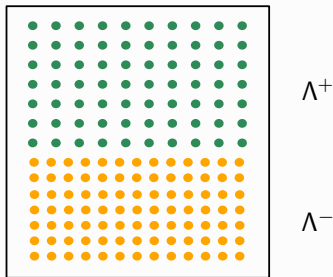
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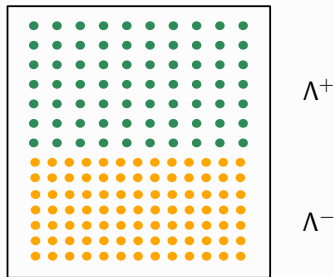


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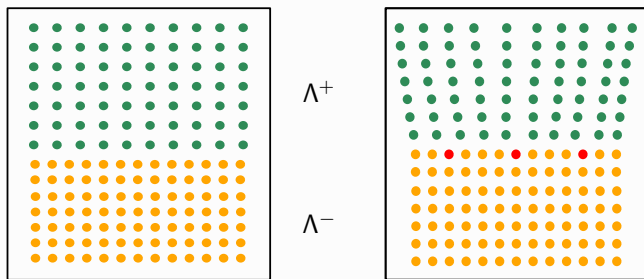
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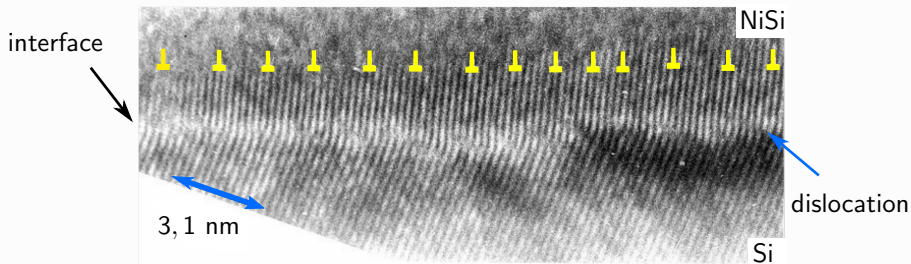
Semi-coherent interface: small dilation $\alpha \approx 1$.

Equilibrium: Λ^+ has lower density than $\Lambda^- \implies$ **edge dislocations** at interface.



Network of dislocations

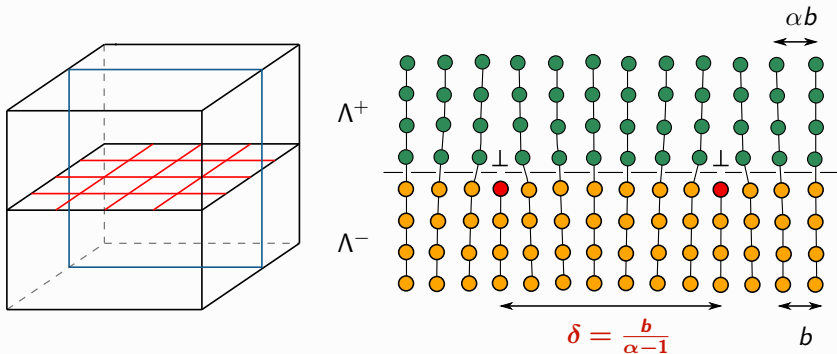
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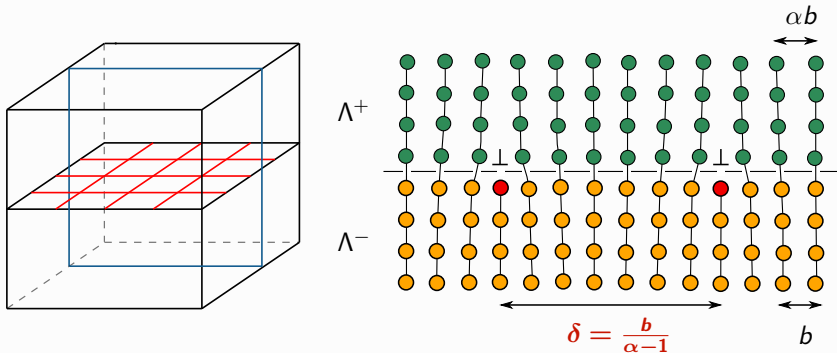
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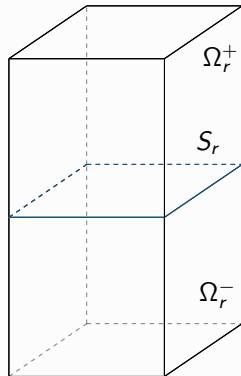
- ▶ analysis of a **semi-discrete model** where dislocations are line defects,
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Semi-discrete line defect model

Reference configuration: $\Omega_r := \Omega_r^- \cup S_r \cup \Omega_r^+$, $r > 0$,

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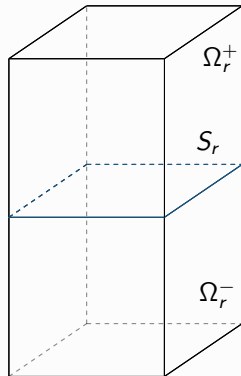


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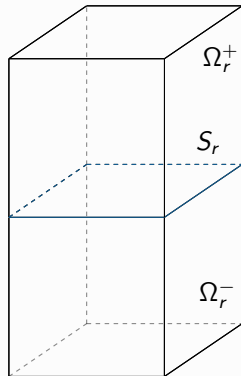
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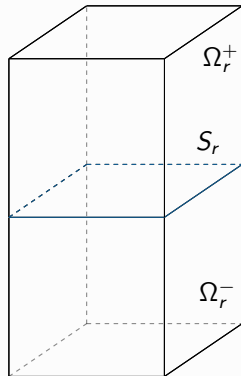
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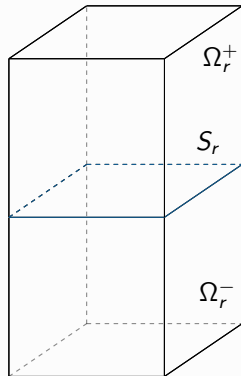
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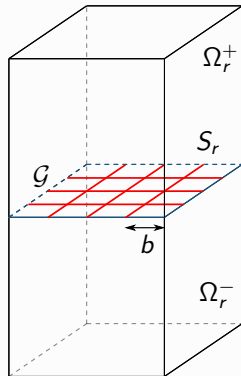
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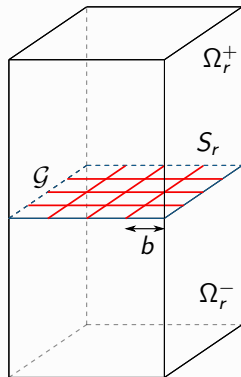
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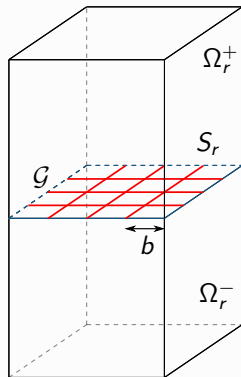
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Müller, Palombaro. *Calculus of Variations and Partial Differential Equations* (2008, 2013).

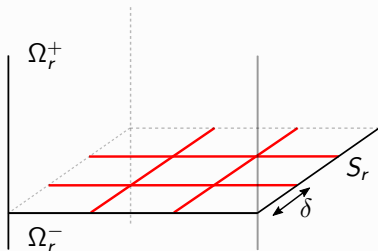
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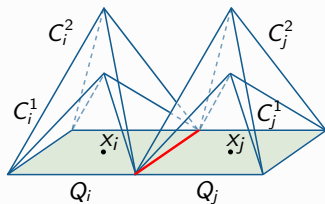
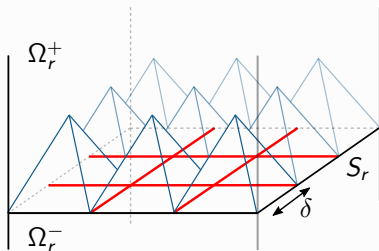
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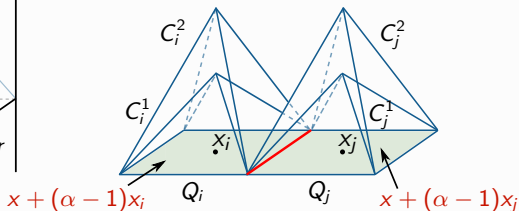
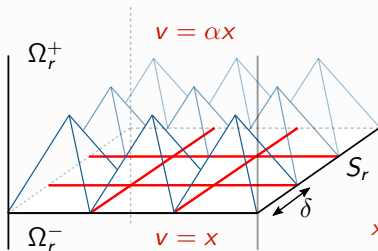
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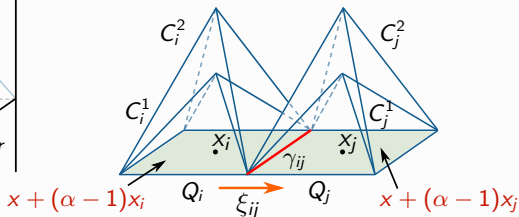
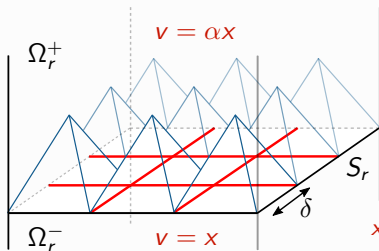
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Induced dislocations: $\text{Curl } \beta = - \sum_{i,j} \xi_{ij} \otimes \dot{\gamma}_{ij} d\mathcal{H}^1 \llcorner \gamma_{ij}$ with

- ▶ $\gamma_{ij} := Q_i \cap Q_j$ admissible dislocation curve ($\alpha = 1 + 1/n \implies \delta = nb$)
- ▶ $\xi_{ij} := (\alpha - 1)(x_j - x_i) \in \pm b\{e_1, e_2\}$ Burgers vector



Upper bound construction

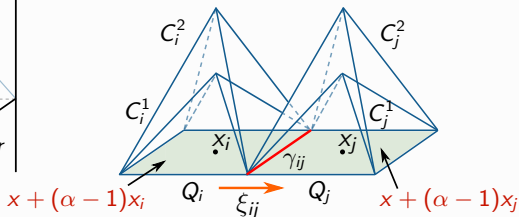
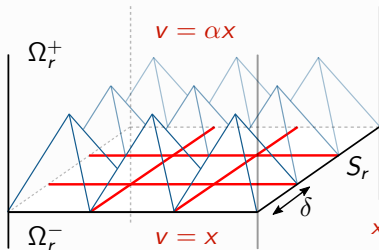
Goal: define a square array of edge dislocations with spacing $\delta := \frac{b}{\alpha - 1}$.

- ▶ Divide S_r into $(r/\delta)^2$ squares of side δ .
- ▶ Above each Q_i define pyramids C_i^1 (height $\delta/2$) and C_i^2 (height δ).
- ▶ Define deformation $\mathbf{v} \in SBV(\Omega_r; \mathbb{R}^3)$, and strain $\beta := \nabla \mathbf{v}$ (a.c. part of $D\mathbf{v}$).

Induced dislocations: $\text{Curl } \beta = - \sum_{i,j} \xi_{ij} \otimes \dot{\gamma}_{ij} d\mathcal{H}^1 \llcorner \gamma_{ij}$ with

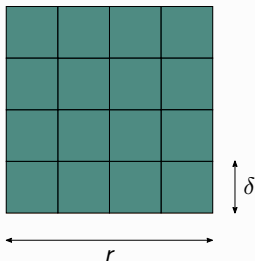
- ▶ $\gamma_{ij} := Q_i \cap Q_j$ admissible dislocation curve ($\alpha = 1 + 1/n \implies \delta = nb$)
- ▶ $\xi_{ij} := (\alpha - 1)(x_j - x_i) \in \pm b\{e_1, e_2\}$ Burgers vector

Energy: in each pyramid is $c = c(\alpha, b, p) \implies E_{\alpha,r} \leq c \frac{r^2}{\delta^2}$ (as $W(\alpha I) = 0$).



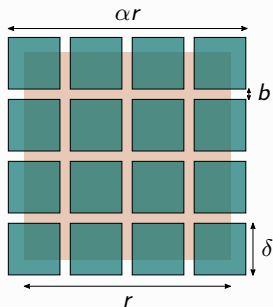
Remarks on the semi-discrete model

Deformed configuration: $v(S_R)$ with v from the upper bound construction



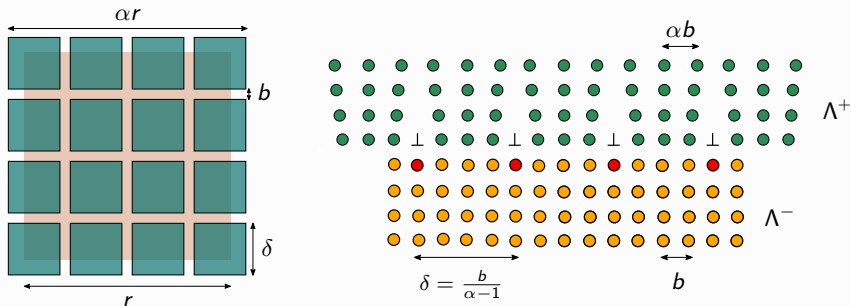
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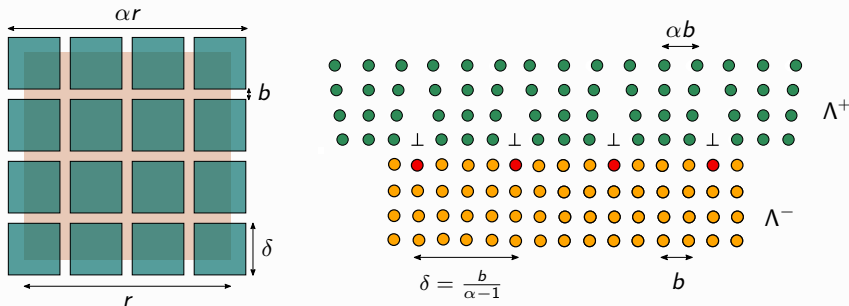
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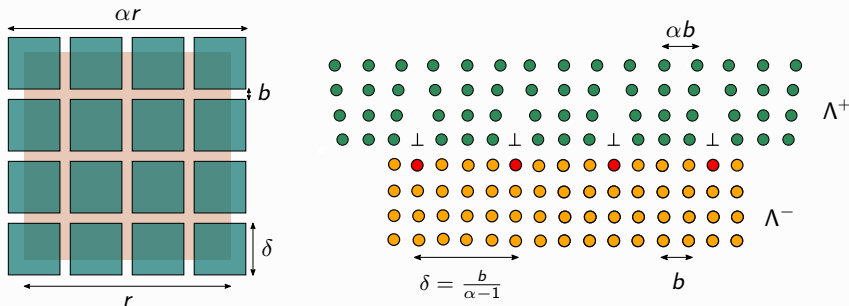


Limitations of the considered model:

- $v(S_r)$ does not match $S_r \implies$ not appropriate for semi-coherent interfaces,

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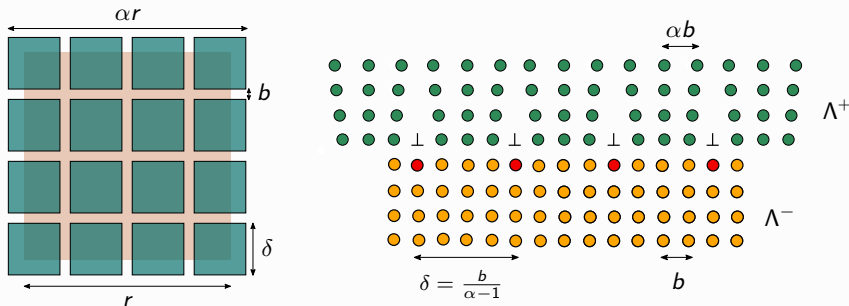


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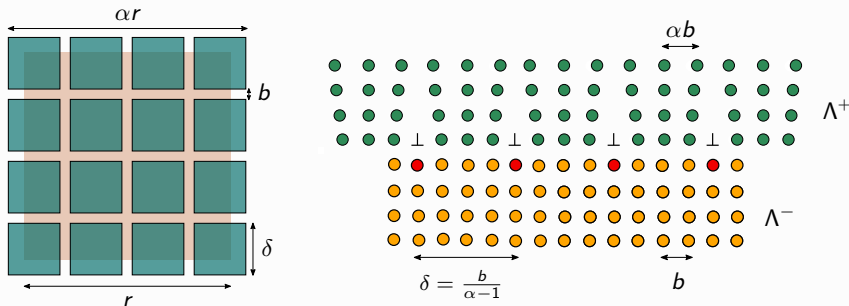
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- ▶ take a smaller overlayer and enforce match at the interface,

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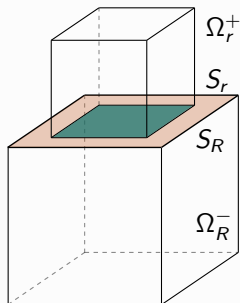
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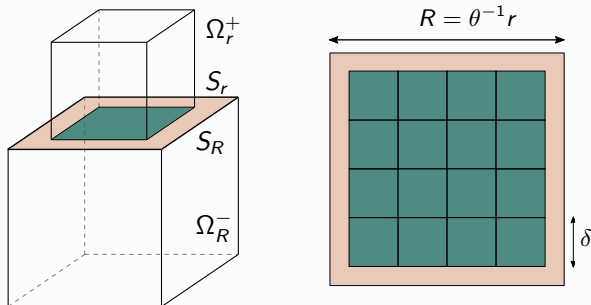
- ▶ take a smaller overlay and enforce match at the interface,
- ▶ introduce a simplified continuum (dislocation density) model to better describe true minimisers.

Heuristic for the continuum model



Reference configuration: $\Omega_{R,r} := \Omega_R^- \cup S_r \cup \Omega_r^+$, with $r := \theta R$, $\theta \in [\alpha^{-1}, 1]$

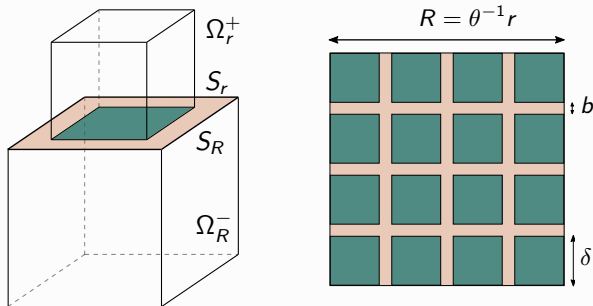
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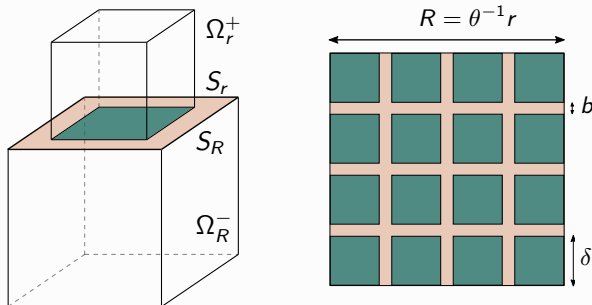
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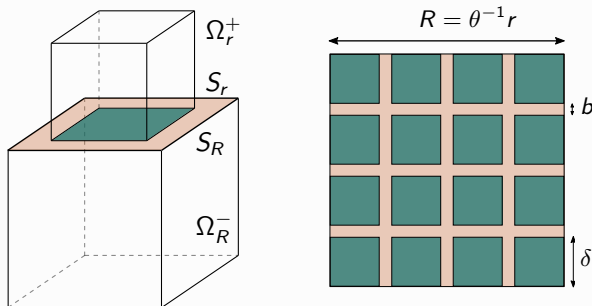


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$$L = 2R \frac{r}{\delta} = \frac{2r^2}{b} (\theta^{-2} - \theta^{-1}) \stackrel{(\theta^{-1} \approx 1)}{\approx} \frac{r^2}{b} (\theta^{-2} - 1) = \frac{1}{b} (R^2 - r^2) = \frac{1}{b} \text{Area Gap}$$

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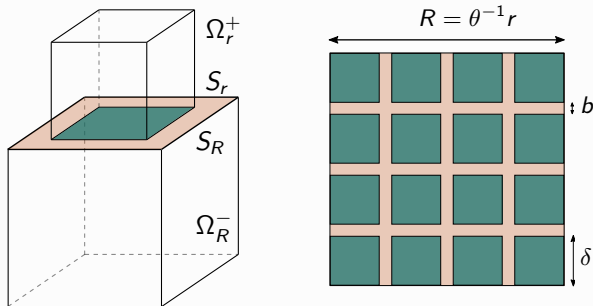


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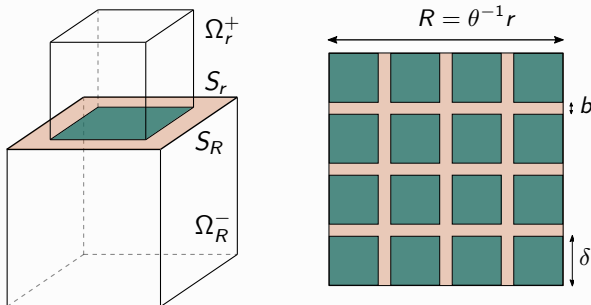
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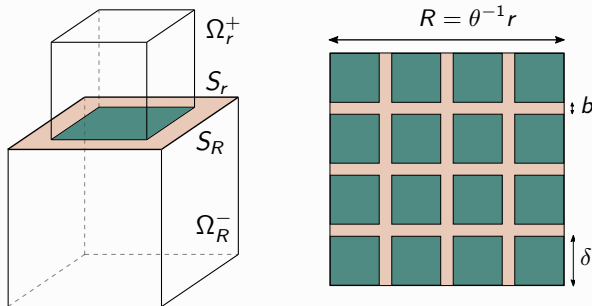
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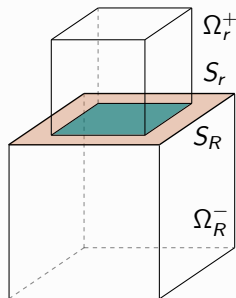
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Hypothesis: Dislocation Energy \propto Dislocation Length. Then optimise over θ .

Continuum model

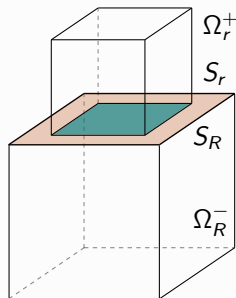
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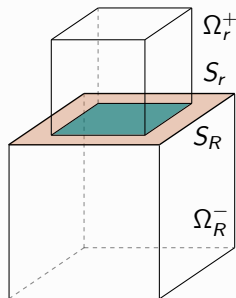


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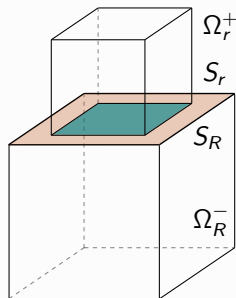
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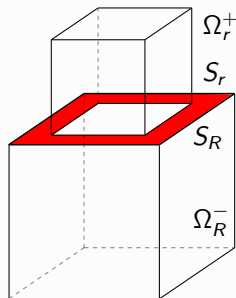
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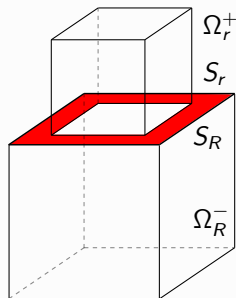
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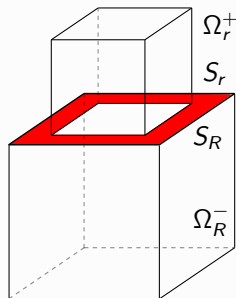
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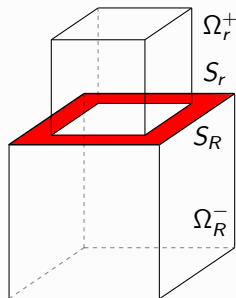
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► $\theta = 1 \implies$ no dislocation energy



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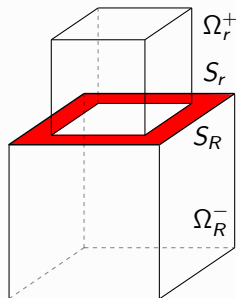
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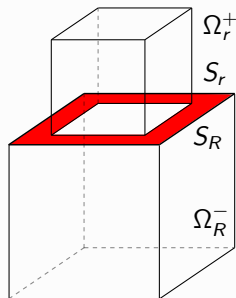
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Asymptotic for $E_{\alpha,R}^{tot}$

Let $\theta_R \in [\alpha^{-1}, 1]$ be a minimiser for $E_{\alpha,R}^{tot}$ and define

$$\mathcal{E}^{el}(R) := \frac{\sigma^2}{\alpha^3 C^{el}} R, \quad \mathcal{E}^{pl}(R) := \sigma R^2 \left(1 - \frac{1}{\alpha^2}\right) - 2 \frac{\sigma^2}{\alpha^3 C^{el}} R.$$

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As $R \rightarrow +\infty$ we have

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Step 4. Write the elastic energy as a polynomial

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where $k_R^{el} := C^{el} + \varepsilon_R > 0$ and $k_R^{el} \rightarrow C^{el}$.

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Step 5. The total energy computed along θ_R is equal to

$$E_{\alpha,R}^{tot}(\theta_R) = k_R^{el} R^3 \theta_R^3 (\theta_R^{-1} - \alpha)^2 + \sigma R^2 (1 - \theta_R^2) \quad (1.1)$$

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Step 8. Since $\theta_R - \theta_R^m \rightarrow 0$, by using (1.1), minimality, and computing $P_{R,k}(\theta_R^m)$, we have the thesis

$$E_{\alpha,R}^{tot}(\theta_R) = \underbrace{\frac{\sigma^2}{\alpha^3 C^{el}} R}_{\text{Elastic}} + \underbrace{\sigma R^2 (1 - \alpha^{-2}) - 2 \frac{\sigma^2}{\alpha^3 C^{el}} R}_{\text{Plastic}} + O(R).$$

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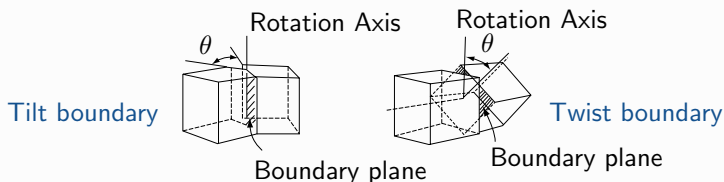
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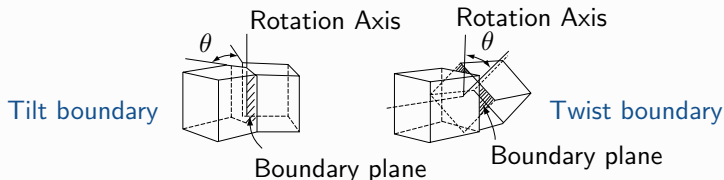
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- ▶ **Grain boundaries**, the misfit between the crystal lattices are described by rotations rather than dilations.

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- ▶ **Optimal geometry** for the dislocation net (square vs hexagonal)

Koslowski, Ortiz (2004)



Presentation Plan

① Geometric Patterns of Dislocations

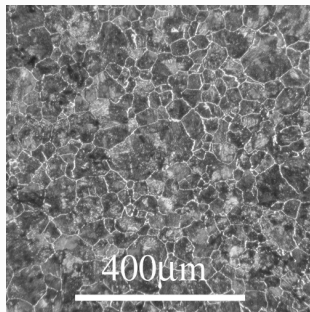
- ▶ Dislocations
- ▶ Semi-coherent interfaces
- ▶ Linearised polycrystals

② Microgeometries in Composites

- ▶ Critical lower integrability
- ▶ Convex integration
- ▶ Proof of our main result

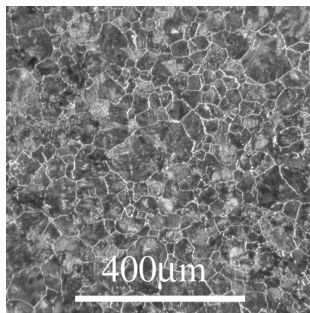
Polycrystals

Polycrystal: formed by many grains, having the **same** lattice structure, mutually rotated \implies interface misfit at **grain boundaries**.



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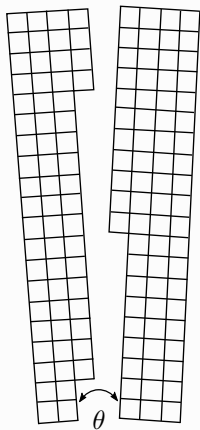
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Goal: obtain polycrystalline structures as minimisers of some energy functional.
F., Palombaro, Ponsiglione. *Linearised Polycrystals from a 2D System of Edge Dislocations*. Preprint (2017)

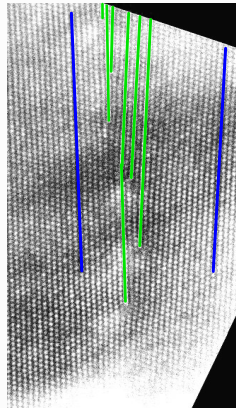
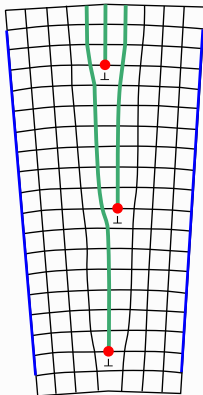
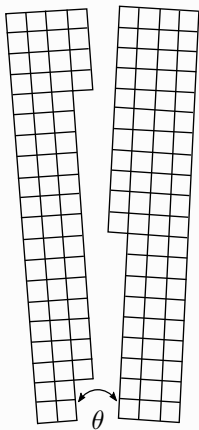
Tilt grain boundaries

Tilt boundary: small angle rotation θ between grains



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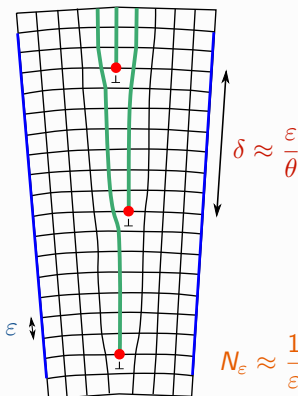
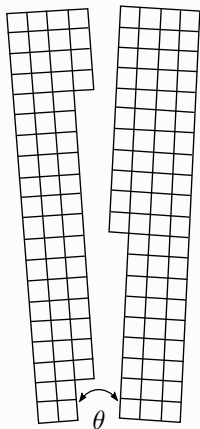
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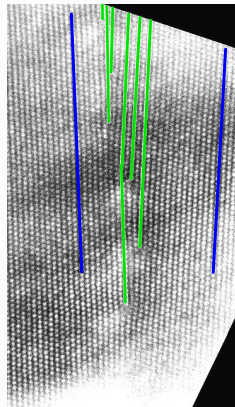
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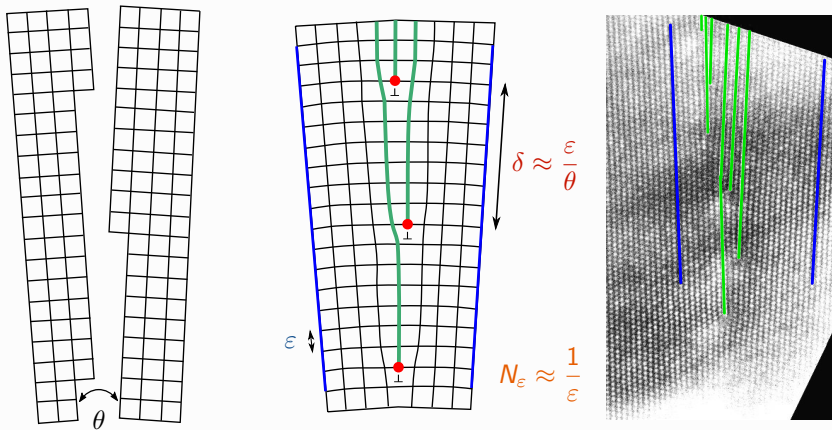
$$N_{\epsilon} \approx \frac{1}{\epsilon}$$



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Porter, Easterling. CRC Press (2009) - Gottstein. Springer (2013)

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Setting: consider a 2D system of N_ε edge dislocations, where $\varepsilon > 0$ is the lattice spacing and

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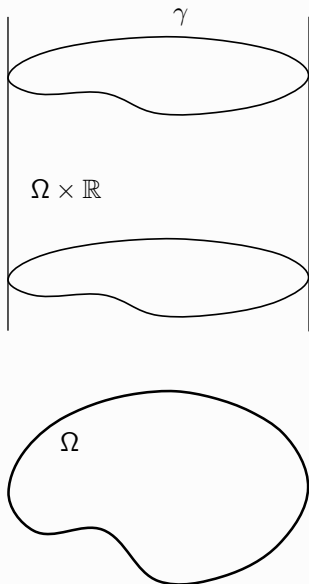
Linearised polycrystals: our energy regime will imply

$$N_\varepsilon \ll \frac{1}{\varepsilon}$$

\implies we have less dislocations than tilt grain boundaries. However we still obtain polycrystalline minimisers, but with grains rotated by an infinitesimal angle $\theta \approx 0$.

Setting (linearised planar elasticity)

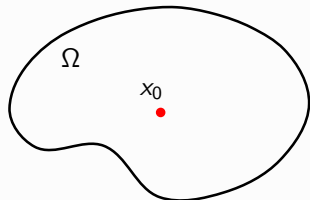
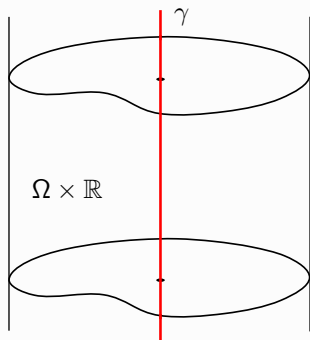
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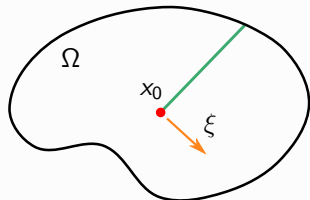
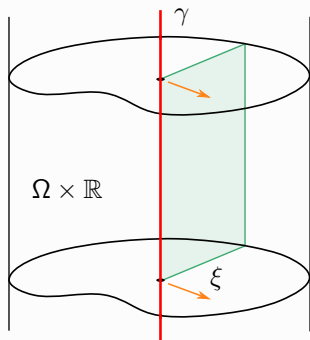


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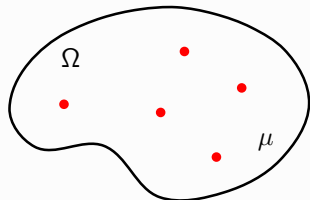
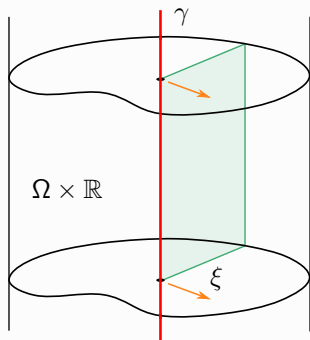
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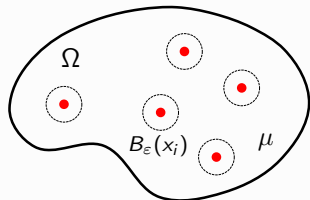
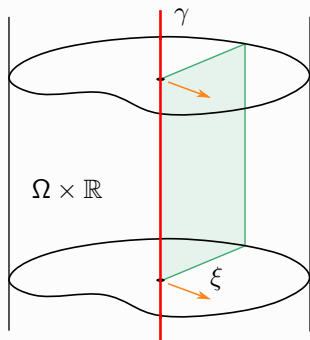
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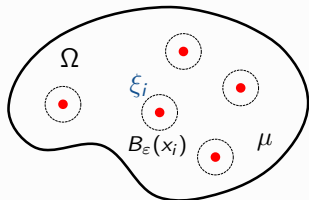
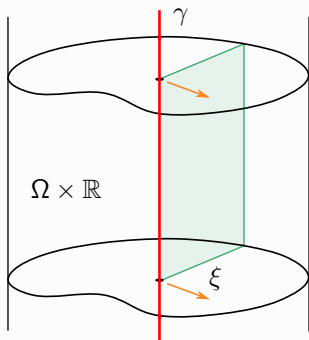
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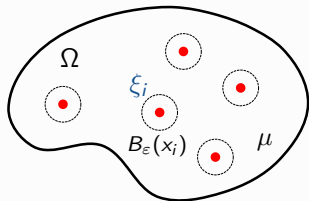
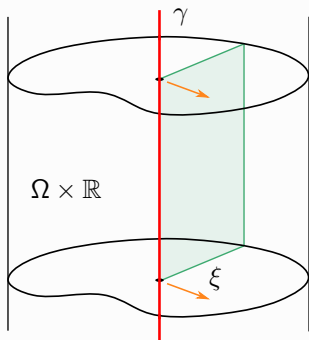
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Linearised Energy: $\mathbb{C}F : F \sim |F^{\text{sym}}|^2$, then

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Self-energy: is of order $|\log \varepsilon|$ and concentrated in a small region around B_ε .

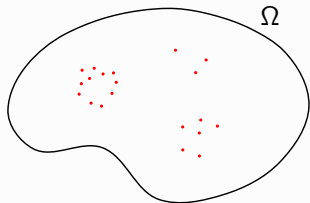
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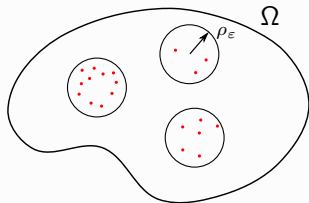
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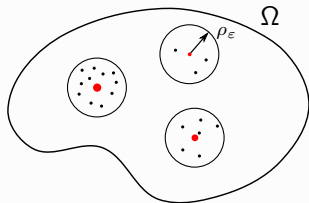
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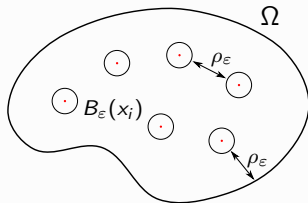
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$$|x_i - x_j| > 2\rho_\varepsilon, \quad \text{dist}(x_k, \partial\Omega) > \rho_\varepsilon.$$



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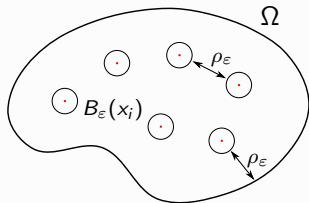
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Admissible dislocations: finite sums of Dirac masses

$$\mu := \sum_{i=1}^N \xi_i \delta_{x_i}, \quad \xi_i \in \mathbb{S},$$

with $\mathbb{S} := \text{Span}_{\mathbb{Z}} \mathcal{S}$ set of multiple Burgers vectors, and

$$|x_i - x_j| > 2\rho_\varepsilon, \quad \text{dist}(x_k, \partial\Omega) > \rho_\varepsilon.$$



Hypothesis on HC Radius: as $\varepsilon \rightarrow 0$

► $\rho_\varepsilon/\varepsilon^s \rightarrow \infty, \quad \forall s \in (0, 1),$ (HC contains almost all the self-energy)

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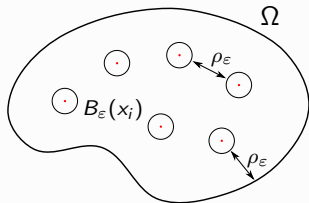
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(HC contains almost all the self-energy)

► $N_\varepsilon \rho_\varepsilon^2 \rightarrow 0.$

(Measure of HC region vanishes)

Energy regimes

Energy scaling: each dislocation accounts for $|\log \varepsilon| \implies$ relevant scaling is

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► $N_\varepsilon \gg |\log \varepsilon| \leadsto$ Super-critical regime

F., Palombaro, Ponsiglione. *Linearised Polycrystals from a 2D System of Edge Dislocations*.
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Candidate Γ -limit

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Density φ : the self-energy for a single dislocation core $\xi \delta_0$ is

$$\psi(\xi) := \lim_{\varepsilon \rightarrow 0} \frac{1}{|\log \varepsilon|} \min_{\beta} \left\{ \int_{B_1 \setminus B_\varepsilon} \mathbb{C}\beta : \beta \, dx : \text{“Curl } \beta = \xi \delta_0 \text{”} \right\}.$$

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Define $\varphi : \mathbb{R}^2 \rightarrow [0, \infty)$ as the relaxation of ψ (splitting multiple dislocations)

$$\varphi(\xi) := \min \left\{ \sum_{i=1}^M \lambda_i \psi(\xi_i) : \xi = \sum_{i=1}^M \lambda_i \xi_i, M \in \mathbb{N}, \lambda_i \geq 0, \xi_i \in \mathbb{S} \right\}.$$

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Theorem (F., Palombaro, Ponsiglione '17)

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Remark:

- ▶ S and A live on two different scales with $S \ll A \implies$ terms in \mathcal{F} decoupled.

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Remark:

- ▶ S and A live on two different scales with $S \ll A \implies$ terms in \mathcal{F} decoupled.
- ▶ In the critical regime $N_\varepsilon \approx |\log \varepsilon|$ we have $S \approx A$ and $\text{Curl}(S + A) = \mu$.

Compactness of the measures

Let $\mu_n := \sum_{i=1}^{M_n} \xi_{n,i} \delta_{x_{n,i}}$ and “ $\text{Curl } \beta_n = \mu_n$ ”. We show that

$$\frac{1}{N_{\varepsilon_n}} |\mu_n|(\Omega) = \frac{1}{N_{\varepsilon_n}} \sum_{i=1}^{M_n} |\xi_{n,i}| \leq C, \quad (1.2)$$

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Compactness of the strains

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$$CN_{\varepsilon_n} |\log \varepsilon_n| \geq CE_{\varepsilon_n}(\mu_n, \beta_n) \geq C \int_{\Omega} |\beta_n^{\text{sym}}|^2 dx$$

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Skew Part: since “ $\text{Curl } \beta_n = \mu_n$ ” we can apply the generalised **Korn inequality**:

$$\int_{\Omega} |\beta_n^{\text{skew}}|^2 dx \leq C \left(\int_{\Omega} |\beta_n^{\text{sym}}|^2 dx + (|\mu_n|(\Omega))^2 \right) \quad (\text{Gen. Korn})$$

$$(N_{\varepsilon} \gg |\log \varepsilon|)$$

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$$\begin{aligned} \int_{\Omega} |\beta_n^{\text{skew}}|^2 dx &\leq C \left(\int_{\Omega} |\beta_n^{\text{sym}}|^2 dx + (|\mu_n|(\Omega))^2 \right) && \text{(Gen. Korn)} \\ &\leq C \left(\sqrt{N_{\varepsilon_n} |\log \varepsilon_n|} + N_{\varepsilon_n}^2 \right) \leq CN_{\varepsilon_n}^2 && (N_{\varepsilon} \gg |\log \varepsilon|) \end{aligned}$$

so that $\frac{\beta_n^{\text{skew}}}{N_{\varepsilon_n}} \rightharpoonup A.$

Garroni, Leoni, Ponsiglione. *Gradient theory for plasticity via homogenization of discrete dislocations.*
J. Eur. Math. Soc. (JEMS) (2010)

Adding boundary conditions

Dirichlet type BC: at level $\varepsilon > 0$ fix a boundary condition $g_\varepsilon: \Omega \rightarrow \mathbb{M}^{2 \times 2}$ s.t.

$$\frac{g_\varepsilon^{\text{sym}}}{\sqrt{N_\varepsilon |\log \varepsilon|}} \rightharpoonup g_S, \quad \frac{g_\varepsilon^{\text{skew}}}{N_\varepsilon} \rightharpoonup g_A.$$

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Remark: $\beta_\varepsilon^{\text{sym}} \ll \beta_\varepsilon^{\text{skew}} \implies$ BC pass to the limit only for A .

Minimising \mathcal{F}_{BC} with piecewise constant BC

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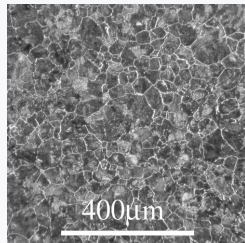
Polycrystals as energy minimisers

Theorem (F., Palombaro, Ponsiglione '17)

Given a *piecewise constant boundary condition* g_A , there exists a *piecewise constant* minimiser of $\mathcal{F}_{\text{BC}}(\mu, 0, A)$

$$A = \sum_{k=1}^M A_k \chi_{E_k},$$

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We interpret A as a *linearised polycrystal*.



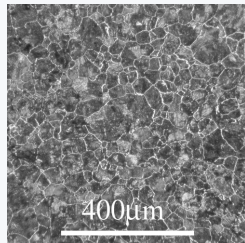
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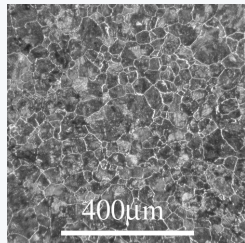
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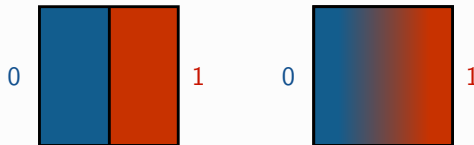
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Essential: that the boundary condition is piecewise affine on the **whole** $\partial\Omega$.



Idea of the proof

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Proof: let \tilde{u} be a minimiser for (1.3). By anisotropic Coarea Formula

$$\int_{\Omega} \varphi \left(\frac{dD\tilde{u}}{d|D\tilde{u}|} \right) d|D\tilde{u}| = \int_{\mathbb{R}} \text{Per}_{\varphi}(\{x \in \Omega : \tilde{u}(x) > t\}) dt,$$

we can select the levels with minimal perimeter. This defines the Caccioppoli partition.

Comparison with classical Read-Shockley formula

Read-Shockley formula: Elastic energy = $E_0\theta(1 + |\log \theta|)$.

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Taylor. *Crystalline variational problems*. Bull. Amer. Math. Soc. (1978).

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- ▶ Supercritical regime analysis starting from a **non-linear energy**?
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Presentation Plan

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$\Omega \subset \mathbb{R}^2$ bounded open domain. A map $\sigma \in L^\infty(\Omega; \mathbb{M}^{2 \times 2})$ is **uniformly elliptic** if

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Study the gradient integrability of distributional solutions $u \in W^{1,1}(\Omega)$ to

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when

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Application to composites:

- ▶ Ω is a section of a **composite conductor** obtained by mixing two materials with **conductivities** σ_1 and σ_2 ,
- ▶ the **electric field** ∇u solves (2.1),
- ▶ concentration of ∇u in relation to the geometry $\{E_1, E_2\}$.

Astala's Theorem



Theorem (Astala '94)

Let $\sigma \in L^\infty(\Omega; \mathbb{M}^{2 \times 2})$ be uniformly elliptic. There exists exponents $1 < q < 2 < p$ such that if $u \in W^{1,q}(\Omega)$ solves

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Question

Are the exponents q and p optimal among two-phase elliptic conductivities

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Astala's exponents for two-phase conductivities



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Upper exponent optimality



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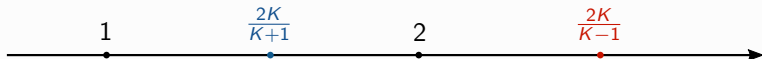
(i) If $\sigma \in L^\infty(\Omega; \{\sigma_1, \sigma_2\})$ and $u \in W^{1, \frac{2K}{K+1}}(\Omega)$ solves

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then $\nabla u \in L_{\text{weak}}^{\frac{2K}{K-1}}(\Omega; \mathbb{R}^2)$.

(ii) There exists $\bar{\sigma} \in L^\infty(\Omega; \{\sigma_1, \sigma_2\})$ and a weak solution $\bar{u} \in W^{1,2}(\Omega)$ to (2.2) with $\sigma = \bar{\sigma}$, satisfying affine boundary conditions and such that $\nabla \bar{u} \notin L^{\frac{2K}{K-1}}(\Omega; \mathbb{R}^2)$.

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Question we address

Is the lower exponent $\frac{2K}{K+1}$ optimal?

Lower exponent optimality



Theorem (F., Palombaro '17)

Let $\sigma_1 = \text{diag}(1/K, 1/S_1)$, $\sigma_2 = \text{diag}(K, S_2)$ with $K > 1$ and $S_1, S_2 \in [1/K, K]$.
There exist

- ▶ coefficients $\sigma_n \in L^\infty(\Omega; \{\sigma_1; \sigma_2\})$,
- ▶ exponents $p_n \in \left[1, \frac{2K}{K+1}\right]$,
- ▶ functions $u_n \in W^{1,1}(\Omega)$ such that $u_n(x) = x_1$ on $\partial\Omega$,

such that

$$\begin{aligned} \operatorname{div}(\sigma_n \nabla u_n) &= 0, \\ \nabla u_n &\in L_{\text{weak}}^{p_n}(\Omega; \mathbb{R}^2), \quad p_n \rightarrow \frac{2K}{K+1}, \quad \nabla u_n \notin L^{\frac{2K}{K-1}}(\Omega; \mathbb{R}^2). \end{aligned}$$

F., Palombaro. Calculus of Variations and Partial Differential Equations (2017)

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① Geometric Patterns of Dislocations

- ▶ Dislocations
- ▶ Semi-coherent interfaces
- ▶ Linearised polycrystals

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- ▶ Critical lower integrability
- ▶ Convex integration
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Solving differential inclusions

Theorem (Approximate solutions for two phases)

Let $A, B \in \mathbb{M}^{2 \times 2}$, $C := \lambda A + (1 - \lambda)B$ with $\lambda \in [0, 1]$, and $\delta > 0$. Assume that

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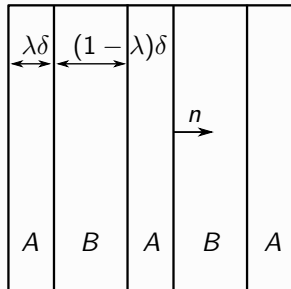
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Müller. *Variational models for microstructure and phase transitions*.

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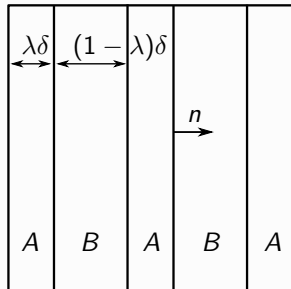
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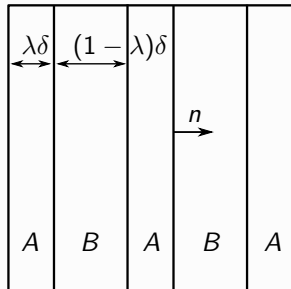
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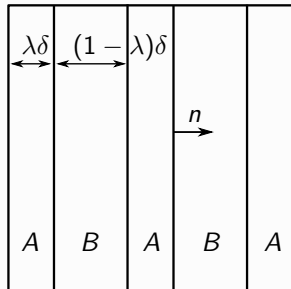
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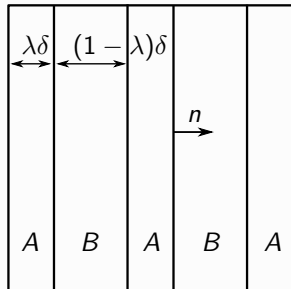
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Müller. *Variational models for microstructure and phase transitions*.

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$$\frac{1}{|\Omega|} \int_\Omega |\nabla f_\delta|^p dx = \int_{\mathbb{M}^{2 \times 2}} |M|^p d\nu_\delta(M).$$

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Let $C = \lambda A + (1 - \lambda)B$ with $\lambda \in [0, 1]$ and $\text{rank}(B - A) = 1$. Let $f: \Omega \rightarrow \mathbb{R}^2$ such that $f(x) = Cx$ on $\partial\Omega$,

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Let $\nu = \sum_{i=1}^N \lambda_i \delta_{A_i}$ be a laminate of finite order, s.t.

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for $f: \Omega \rightarrow \mathbb{R}^2$ and an appropriate target set $T \subset \mathbb{M}^{2 \times 2}$.

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These methods were developed for isotropic conductivities $\sigma \in L^\infty(\Omega; \{KI, \frac{1}{K}I\})$.

The adaptation to our case is non-trivial because of the lack of symmetry of the target set T , due to the anisotropy of σ_1 and σ_2 .

Astala, Faraco, Székelyhidi. *Convex integration and the L^p theory of elliptic equations*.

Ann. Scuola Norm. Sup. Pisa Cl. Sci. (2008)

Rewriting the PDE as a differential inclusion

Let $K > 1$, $S_1, S_2 \in [1/K, K]$ and define

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A function $u \in W^{1,1}(\Omega)$ is solution to

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The sets of conformal linear maps and anti-conformal linear maps are

$$E_0 := \{(z, 0) : z \in \mathbb{C}\} \quad (\text{Conformal maps})$$

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Targets in conformal coordinates

Conformal coordinates: Let $A \in \mathbb{M}^{2 \times 2}$. Then $A = (a_+, a_-)$ for $a_+, a_- \in \mathbb{C}$, defined by

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Target sets in conformal coordinates are

$$T_1 = \{(a, d_1(\bar{a})) : a \in \mathbb{C}\}, \quad T_2 = \{(a, -d_2(\bar{a})) : a \in \mathbb{C}\},$$

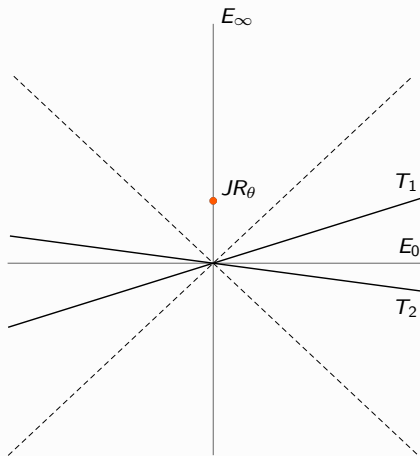
where the operators $d_j: \mathbb{C} \rightarrow \mathbb{C}$ are defined as

$$d_j(a) := k \operatorname{Re} a + i s_j \operatorname{Im} a, \quad \text{with} \quad k := \frac{K-1}{K+1} \quad \text{and} \quad s_j := \frac{S_j-1}{S_j+1}.$$

Staircase Laminate (F., Palombaro '17)

Let $\theta \in [0, 2\pi]$, $JR_\theta = (0, e^{i\theta})$.

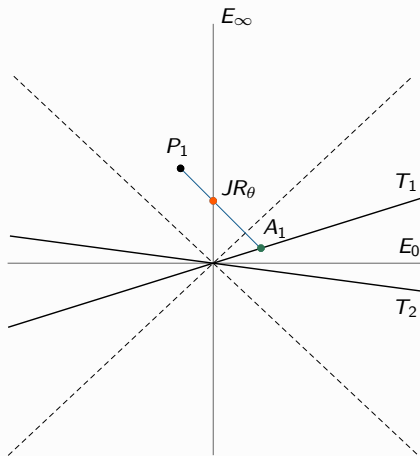
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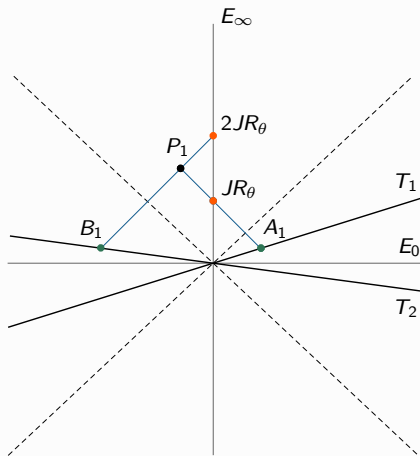
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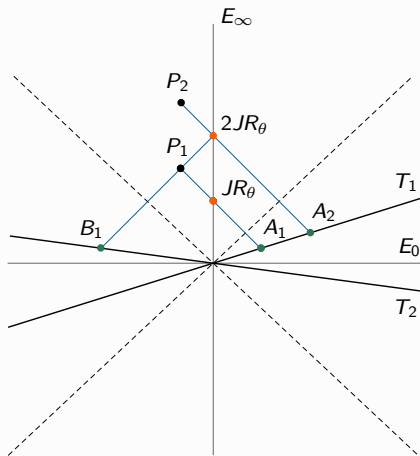


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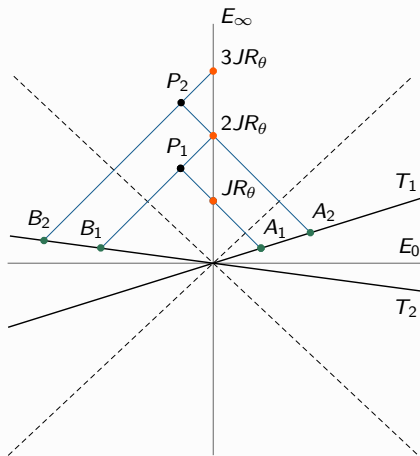


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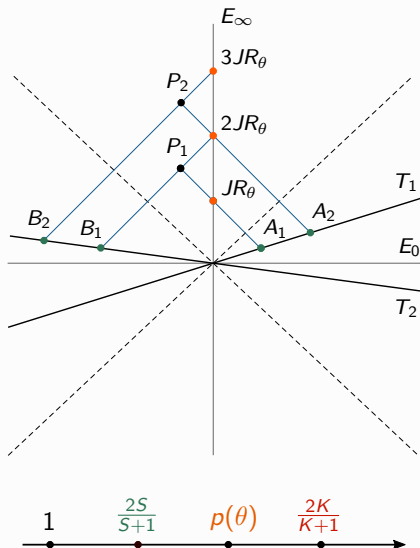
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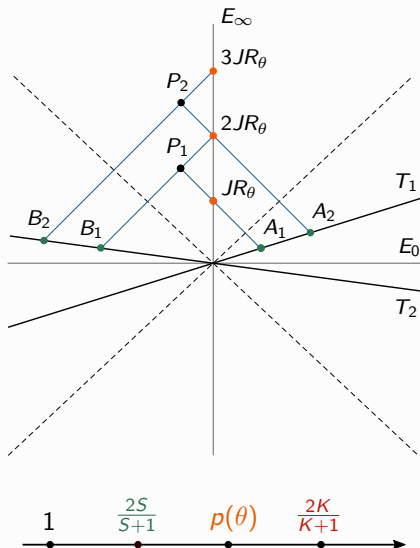
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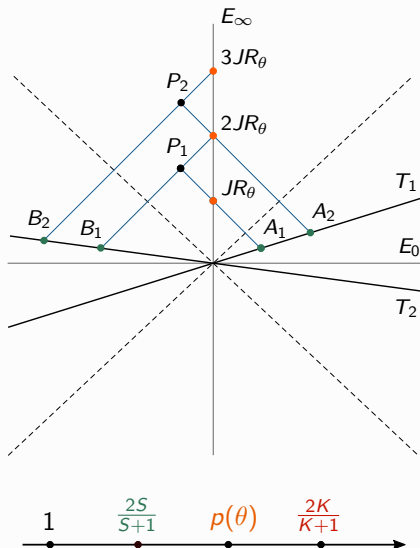
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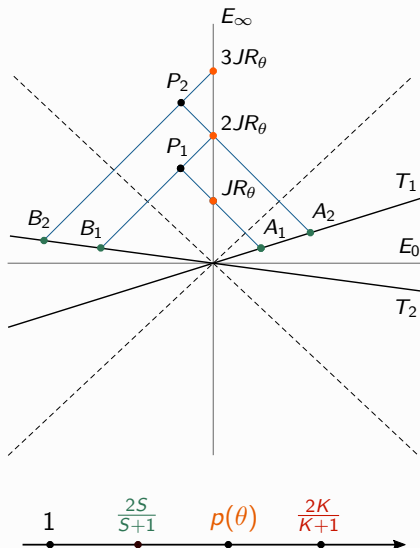
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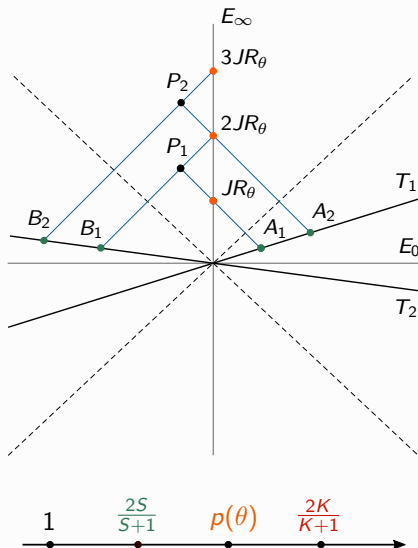
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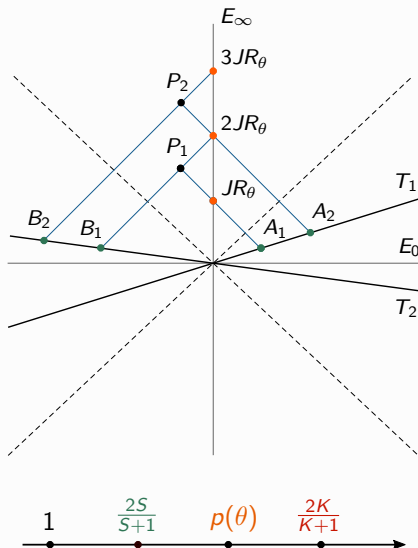
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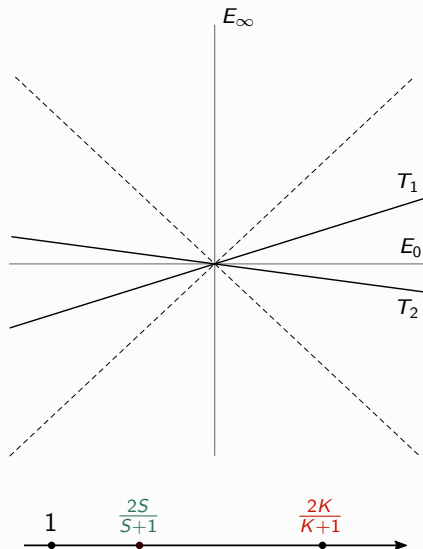
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Remark: barycentre J gives the right growth.



Constructing approximate solutions

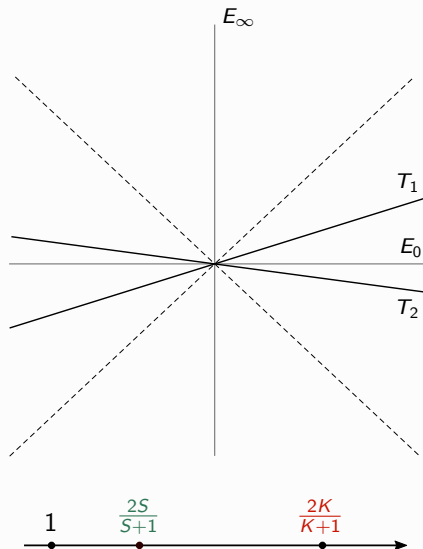
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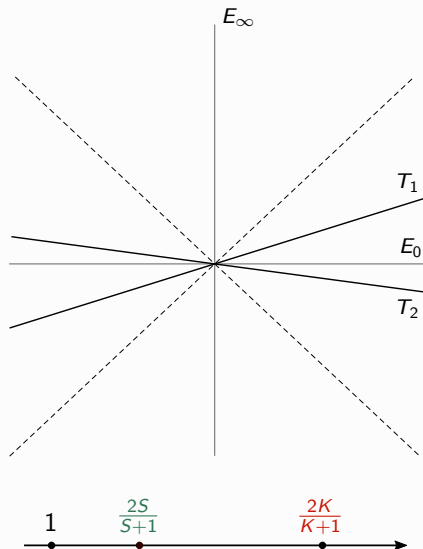
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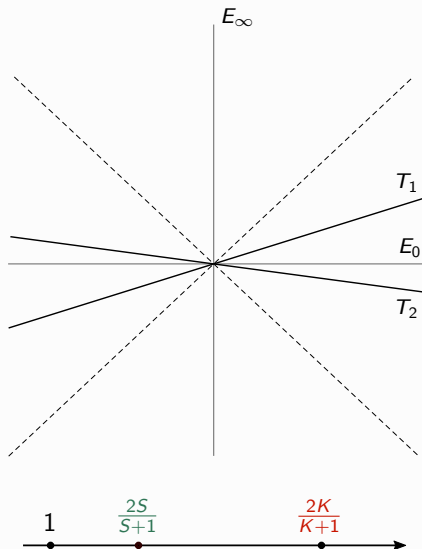
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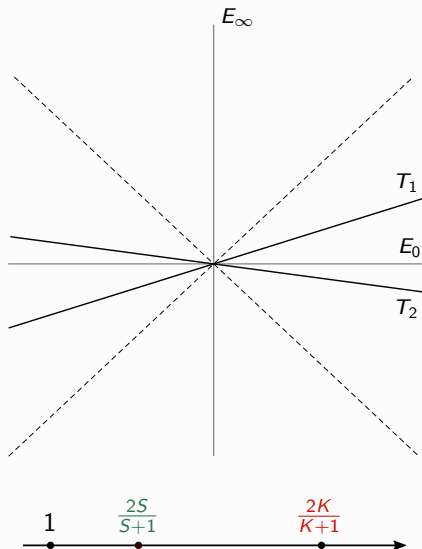
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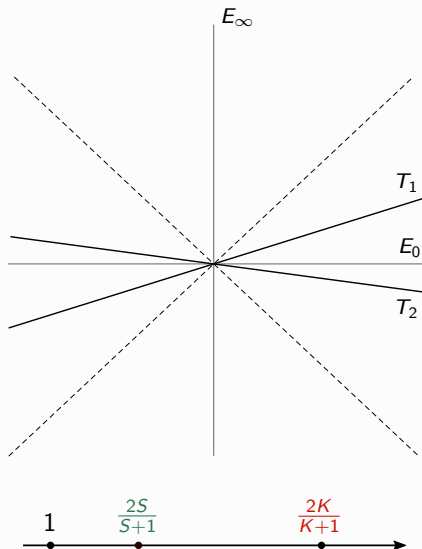


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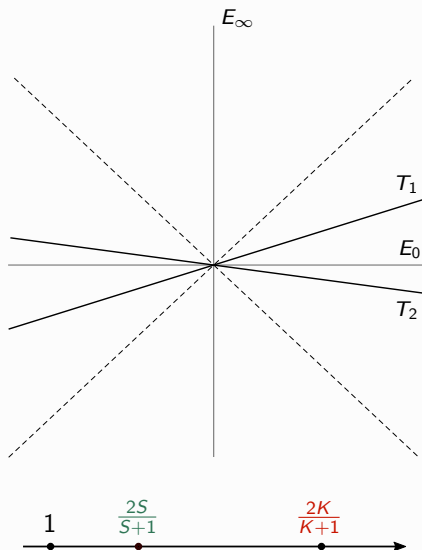
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Idea: alternate one step of the staircase laminate with the convex integration Proposition.



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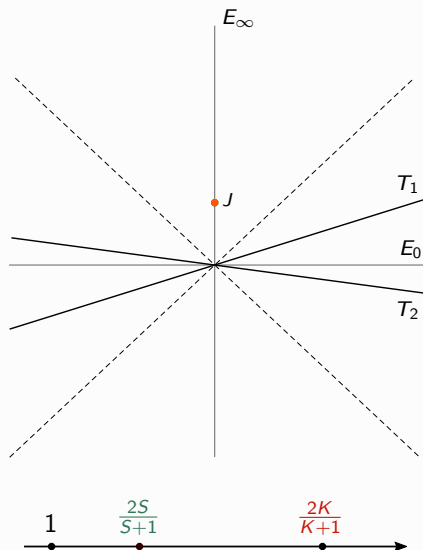
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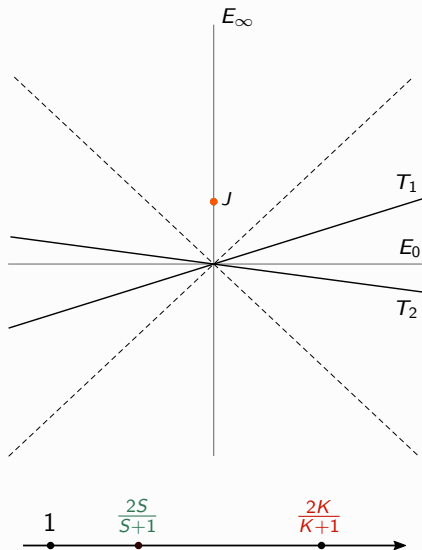


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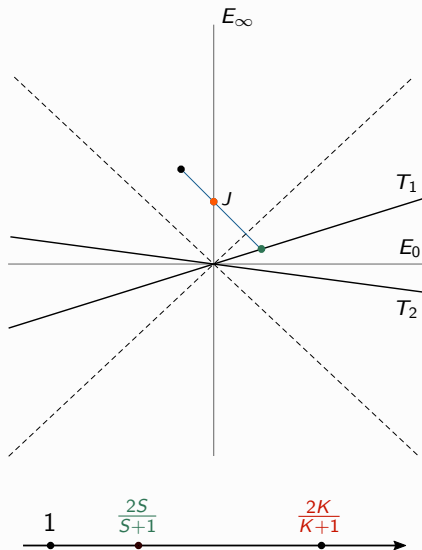


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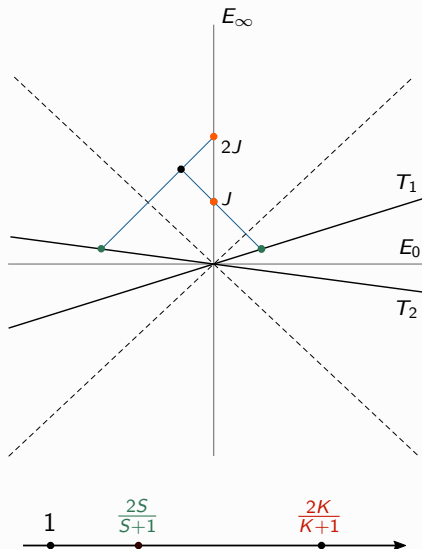


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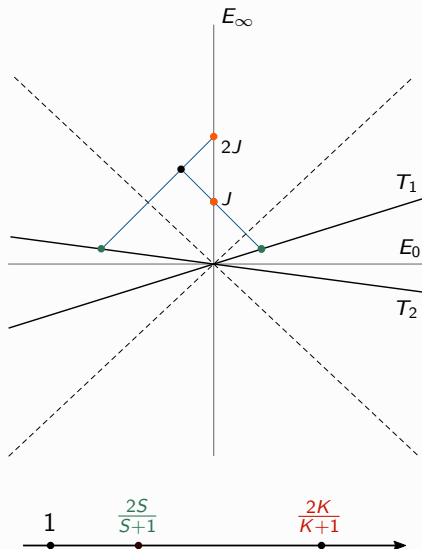
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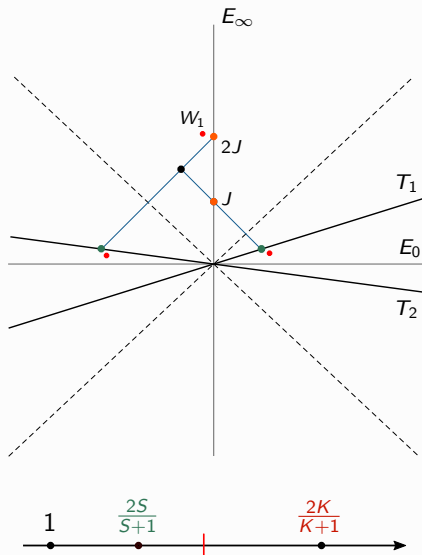
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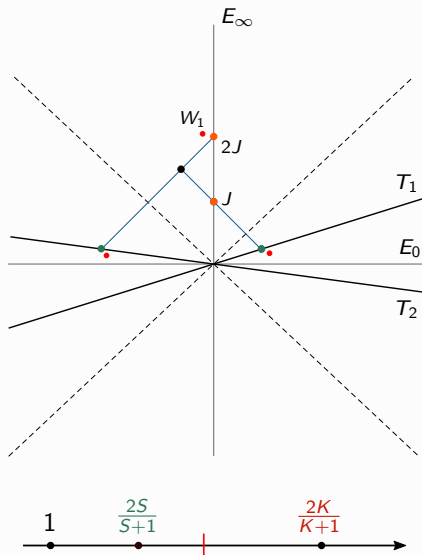
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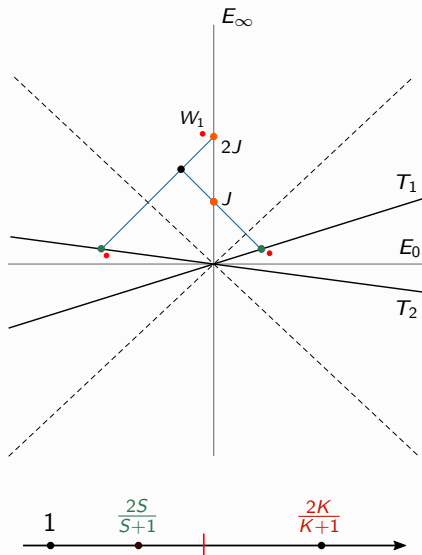
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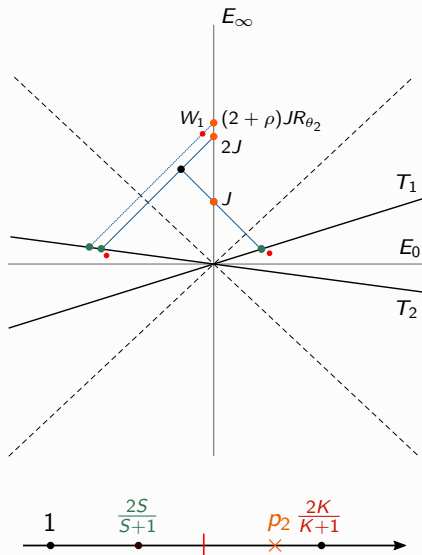
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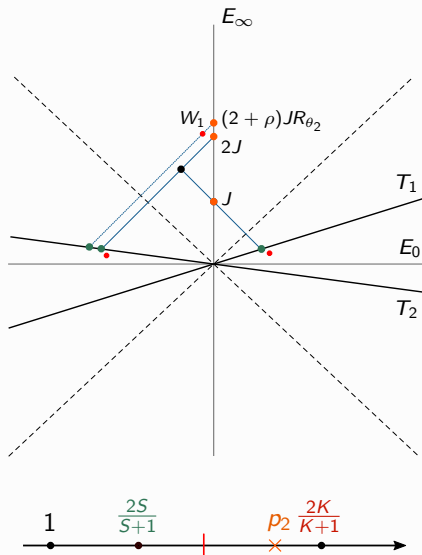
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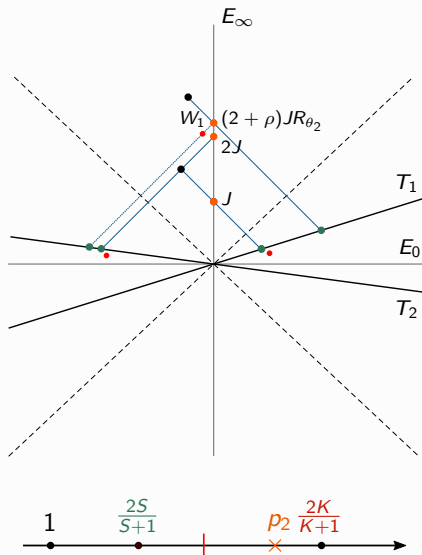
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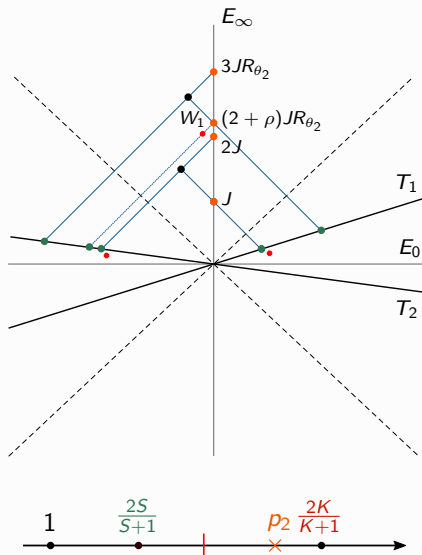
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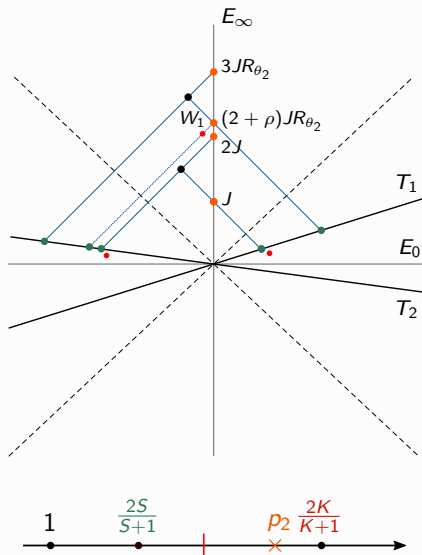
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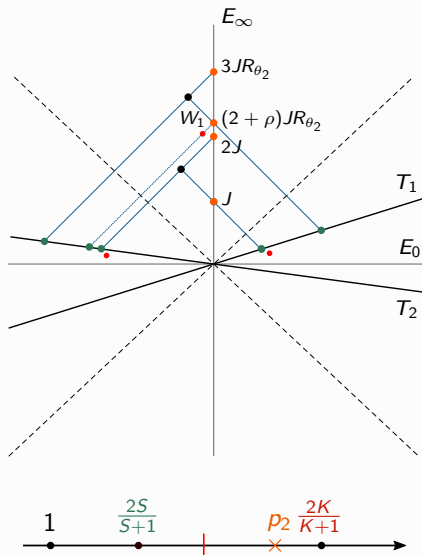
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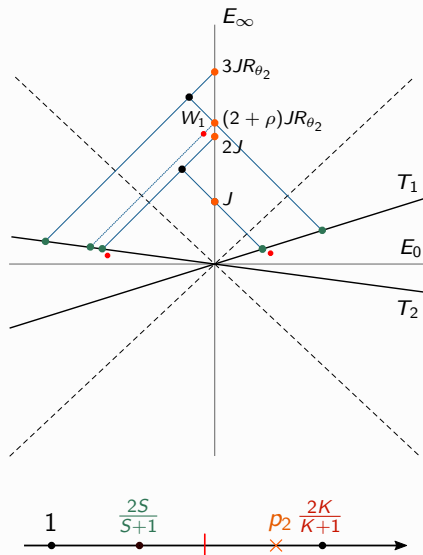
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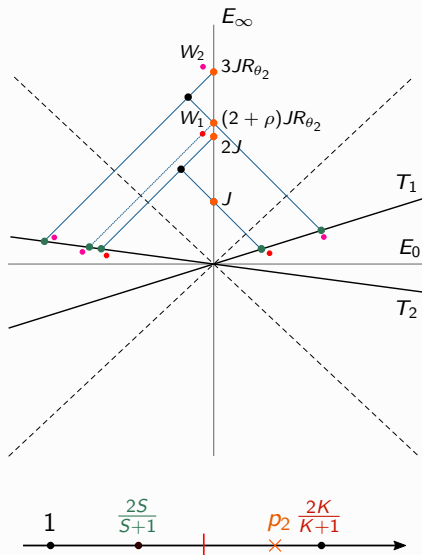
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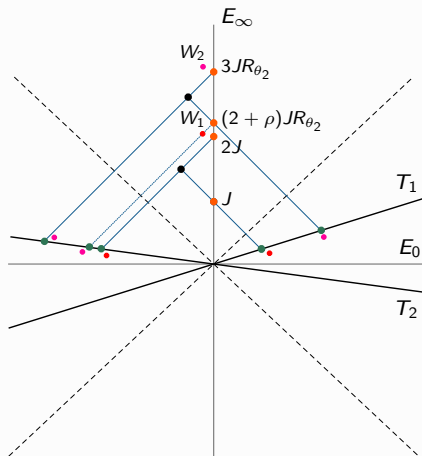
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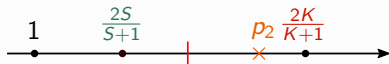
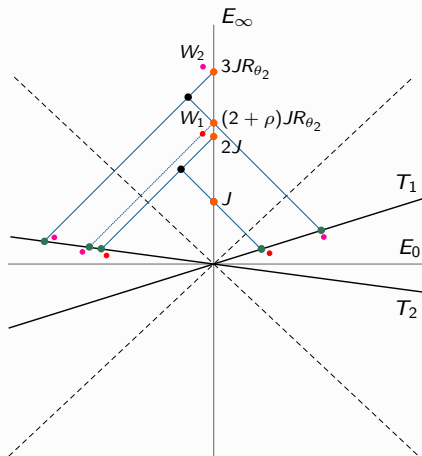
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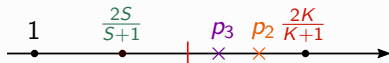
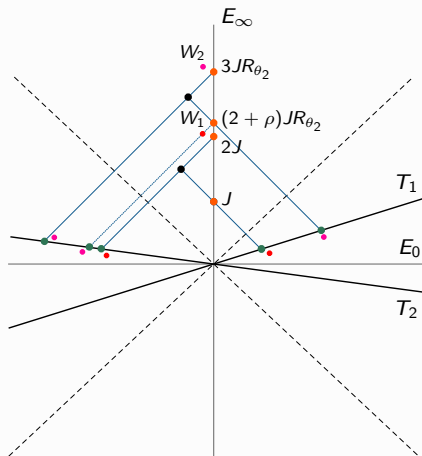
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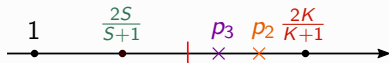
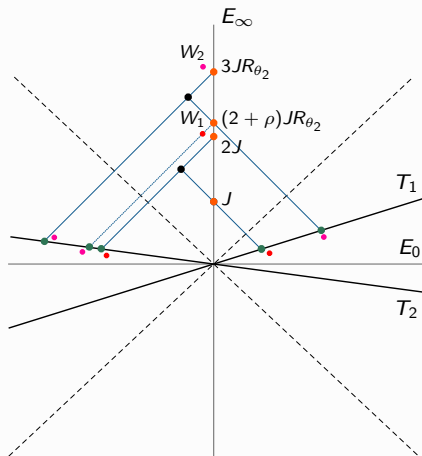
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Iterating: $\rightsquigarrow f_n$ obtained by successive modifications
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- ▶ **Dimension $d \geq 3$?** Even only in the isotropic case $\sigma \in \{KI, K^{-1}I\}$ for $K > 1$.
Main difficulty: Astala's Theorem is missing in higher dimensions.

Thank You!

