

Plasticity and Dislocation Dynamics in a Phase Field Crystal Model

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The critical dynamics of dislocation avalanches in plastic flow is examined using a phase field crystal model. In the model, dislocations are naturally created, without any *ad hoc* creation rules, by applying a shearing force to the perfectly periodic ground state. These dislocations diffuse, interact and annihilate with one another, forming avalanche events. By data collapsing the event energy probability density function for different shearing rates, a connection to interface depinning dynamics is confirmed. The relevant critical exponents agree with mean field theory predictions.

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Materials yield and deform plastically under large external stress. While the yield surface and the plastic flow are well described by various continuum theories [1–3], what happens microscopically during a plastic deformation is still not fully understood. On atomic scales, external stress is carried by localized crystal defects, such as dislocations and disclinations. They are created under stress and interact with each other. Although the properties of individual defects and their interactions are well known [4,5], their collective behavior under external stress is complicated. It gives rise to scale-invariant, power-law-distributed phenomena [6–12], in strong resemblance to the scaling behavior near a critical point. These phenomena include dislocation slip avalanches of a broad range of sizes and acoustic emission with a power-law power spectrum.

Recently, evidence has accumulated that these scaling phenomena reflect an underlying nonequilibrium critical point [6,7], i.e., a point where the nonequilibrium steady state fluctuations are governed by a diverging correlation length [13], as appears to be the case in magnetic materials [14] and even turbulence [15]. Such a critical point would exist at the boundary between two distinct regimes, one of which would be a glassy, activated regime, and the other would be a genuine plastic flow regime. Up to now, experimental and computational data have focused on the glassy regime: For example, Weiss *et al.* have measured the acoustic emission signal from creep deformation experiments on single crystal ice and found that the event size distribution follows a power law over 4 decades [7,8]. Miguel *et al.*'s dislocation dynamics simulations in two dimensions showed event size distributions following a power law, with a rate-dependent cutoff, over approximately 2 decades [9]. Zaiser has reported a data collapse of the event size distribution with different external stresses [6].

The purpose of this Letter is to approach plastic deformation from the flow side of the nonequilibrium critical point, manifested in the strain-rate dependence of the acoustic emission. Importantly, we are able to systemati-

cally vary the strain rate in simulations, and moreover we relate the critical point underlying plastic flow at finite strain rates to the scaling of magnetic domain wall depinning [6,16,17]. We find remarkable agreement between simulations and analytical mean field theory predictions of exponents and, in addition, are able to show that the strain-rate data exhibit collapse. Our results strongly support the critical point picture of plasticity and suggest new experiments. We study dislocation avalanches during plastic flows by using the phase field crystal (PFC) model [18,19]. This approach is well-suited to this problem, because it can be performed at finite temperature, for large systems, and over long time periods. The PFC model describes the dynamics of the local crystalline density field and has been shown to give an excellent account of numerous materials properties including polycrystalline solidification, vacancy diffusion, grain growth, grain boundary energetics, epitaxial growth, fracture [19], grain coarsening [20], elasticity [21], dislocation annihilation, glides, and climb [22], as well as vacancy dynamics [23]. The model can be derived from density-functional theory and extended to the case of binary alloys [24]. In this paper, we augment the model to treat external shearing forces by adding an advective term to the dynamics near the boundary. By adjusting the shearing force and measuring the resulting avalanche statistics, a data collapse is obtained that is consistent with proximity to a domain wall depinning point at finite temperature [6,17].

The model.—The PFC model is given by the free energy density [18,19]

$$f = \frac{\rho}{2}(\nabla^2 + 1)^2\rho + \frac{r}{2}\rho^2 + \frac{\rho^4}{4}, \quad (1)$$

where r is the undercooling and $\rho(\vec{x}, t)$ is the local density. The dynamics associated with this free energy is conservative, relaxational and diffusive, and systematically derivable from density-functional theory [25]. In order to study the plastic response of the PFC model under shear, we add a shearing term to the dynamical equation:

$$\frac{\partial^2 \rho}{\partial t^2} + (\beta) \frac{\partial \rho}{\partial t} = (\alpha)^2 \nabla^2 \frac{\delta F}{\delta \rho} + v(y) \frac{\partial \rho}{\partial x} + \eta, \quad (2)$$

where

$$v(y) = \begin{cases} v_0 e^{-y/\lambda} & \text{for } 0 < y < L_y/2 \\ -v_0 e^{-(L_y-y)/\lambda} & \text{for } L_y/2 < y < L_y \end{cases} \quad (3)$$

is the shearing profile and v_0 is the magnitude of the shearing, λ is the penetration depth, α and β control the range and time scale, respectively, of elastic interactions (phonon excitations) propagating through the medium [21], $F \equiv \int f(\vec{x}) d^d x$ is the total free energy, and η is the thermal noise satisfying the fluctuation-dissipation theorem $\langle \eta(\vec{x}, t) \eta(\vec{x}', t') \rangle = -\epsilon \nabla^2 \delta(\vec{x} - \vec{x}') \delta(t - t')$. Here ϵ is the noise amplitude. It is directly proportional to the temperature $k_B T$. The value of v_0 controls the magnitude of the shearing force; the penetration depth λ controls how deep the shearing force extends into the material. In all simulations we set $\lambda \ll L_y$, so the actual value of λ does not affect our simulation results. The boundary conditions are periodic in x and fixed at $y = 0, L$. The PFC model we used allows propagating sound modes [21] when $\beta = 0.9$; as long as β is nonzero and $O(1)$, we do not expect that our scaling results will be sensitive to its precise value. Values of β that are $O(10)$ would correspond to very overdamped dynamics and could model viscoelastic behavior [21] that is outside the scope of our work.

One of the advantages of using the PFC model is that we do not have to impose any *ad hoc* assumptions about the creation and annihilation of dislocations. Recall that in dislocation dynamics simulations, dislocations are treated as elementary particles and usually only the far field interaction between dislocations is captured. When dislocations get too close to each other (a few atomic spacings), the highly nonlinear interaction between them is not captured and, more importantly, the annihilation of dislocations is not accounted for. The standard practice is then to impose some annihilation rules—dislocations of opposite topological charges annihilate when they get too close to each other [9]. Similarly, dislocations have to be created by hand when the local strain is high. Although these rules are consistent with our physical intuition, particular ways of implementing them are sometimes difficult to justify. However, because the PFC model captures the nonlinear elastic behavior of a crystal, the interaction between dislocations is completely captured. In addition, because the PFC model simulates the atoms in the lattice (the PFC density is periodic in its ground state with peaks representing atoms and troughs interatomic space) but not the dislocations themselves, creation and annihilation of them are also naturally captured as collective excitations of the lattice. No *ad hoc* rules or assumptions have to be imposed.

We solved Eq. (2) in a 2D rectangular domain. The crystal under shear is initially perfectly triangular. As the crystal is sheared, dislocations are created near the fixed boundaries $y = 0, L$ where the stress is higher. They then propagate into the bulk. They interact with each other and

form avalanches. To quantify the avalanche activity, we calculate the total speed of all dislocations in the domain: $\tilde{V}(t) = \sum_{i=1}^{N_{\text{dis}}(t)} |\vec{u}_i|$, where $N_{\text{dis}}(t)$ is the number of dislocations in the system at time t and \vec{u}_i is the velocity of the i th dislocation. This measure is similar to the acoustic emission signal in Weiss *et al.*'s single crystal ice experiments. As dislocations are generated and interact with each other in the domain, in addition to the fast avalanching dynamics, quasistatic structures, such as grain boundaries, can form. These slow dynamics should not be measured because they are really not part of the avalanches. This leads to the distinction between fast-moving and slowly moving dislocations introduced by Miguel *et al.* [9]. In essence, they introduced a cutoff in dislocation speed and measure only dislocations with speed higher than the cutoff. In that way, they tried to retain only the avalanche activities in the acoustic emission signals.

We employed a different method to tackle this problem. Instead of simulating a very large system, with all sorts of dislocation activities, we simulated a moderate size of system with approximately 10 000 atoms. For this system size, dislocation avalanches come and go; i.e., not many dislocations are left in the system after every avalanche. As a result, no grain boundaries, or slow dynamics, are present and we obtain clean avalanche data. It is fair to mention that this method severely limits the system size and thus the resulting avalanche sizes. The system size we chose contains approximately 100 dislocations in the largest avalanche events. The trade-off, which we exploit, is the cleanliness of the avalanche signal and the speed of the resulting simulations. Different methods, such as those we mentioned above, would have to be employed if larger avalanche sizes are desired.

We count the number of nearest neighbors of each atom, n_i , by using the Delaunay triangulation method in computational geometry [26,27]. Because we have $n_i = 6$ for every atom in a perfectly triangular crystal, and because there are no vacancies introduced in the version of the PFC model simulated here (vacancies can be introduced into the PFC model by breaking the up-down symmetry of the PFC free energy, as detailed in Ref. [23]), any atom having $n_i \neq 6$ is sitting next to a dislocation. Because the PFC exhibits emergent rigidity in the region of the phase diagram studied here, these “defect atoms” essentially track the locations of dislocations. Instead of measuring the total sum of dislocation speeds $\tilde{V}(t)$, we then measure the total sum of these defect atoms' speeds:

$$V(t) = \sum_{i=1}^{N(t)} |\vec{v}_i|, \quad (4)$$

where $N(t)$ is the number of defect atoms and \vec{v}_i is the velocity of defect atom i . Note that the velocity of a defect atom is not the velocity of any atom in the system, but the velocity of the dislocation it is tracking. Because the two measures $\tilde{V}(t)$ and $V(t)$ are proportional to each other with the proportionality constant being the mean number of

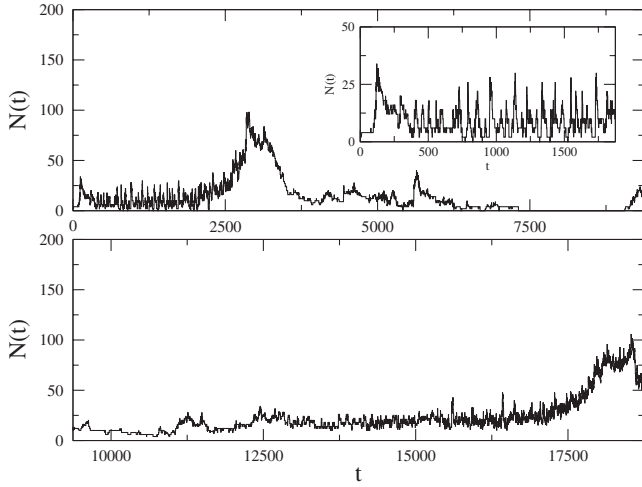


FIG. 1. The number of dislocations in a sheared PFC crystal. Intermittent events with sizes differing in orders of magnitude are observed. Parameters are $(\alpha)^2 = 255$, $\beta = 0.9$, $v_0 = 1.581$, $\rho_0 = 0.3$, $\epsilon = 1.5$, $\lambda = 40.0$, and $r = -0.5$.

defect atoms sitting next to a dislocation, we can use the latter for convenience. Figure 1 shows the typical time dependence of $N(t)$ from a simulation with parameters $dx = 3\pi/8$, $dt = 0.025$, $L_x = L_y = 512$, $(\alpha)^2 = 255$, $\beta = 0.9$, $v_0 = 1.581$, $\rho_0 = 0.3$, $\epsilon = 1.5$, $\lambda = 40.0$, and $r = -0.5$. $N(t)$ changes as dislocations are being created and annihilated. There are intermittent events of creation ranging from a few to 80. Figure 2 shows the acoustic emission signal $V(t)$ in the same simulation. Similar to $N(t)$, the signal ranges from 0 to 400, with intermittent pulses of various sizes. In order to measure the avalanche event size, we introduce a cutoff to the signal $V_{\text{cut}} = 15$ for shear velocity $v_0 = 0.765$. The cutoff is increased propor-

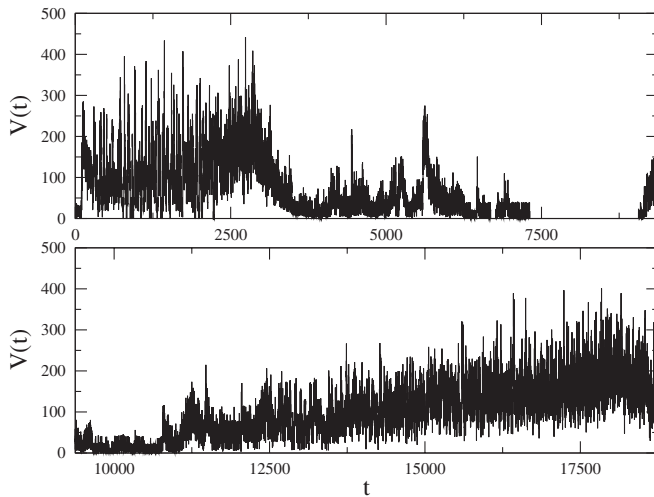


FIG. 2. The total speed of defect atoms in a sheared PFC crystal. Intermittent events with sizes differing in orders of magnitude are observed. Parameters are $(\alpha)^2 = 255$, $\beta = 0.9$, $v_0 = 1.581$, $\rho_0 = 0.3$, $\epsilon = 1.5$, $\lambda = 40.0$, and $r = -0.5$.

tionally to the shear velocity and is primarily used to define the avalanche size, rather than to remove slowly moving dislocations from the analysis. The signal is then partitioned into individual avalanche events. The probability distribution of the event energy

$$E = \int_{t_{\text{begin}}}^{t_{\text{end}}} V^2(t) dt, \quad (5)$$

where t_{begin} and t_{end} are the starting and ending time of the event, respectively, can be measured. For cutoff values small compared to the signal (i.e., $V_{\text{cut}} = 15$, corresponding to the activity of 3–4 dislocations due to thermal creep), the result is insensitive to the cutoff. For each shearing rate, at least 10 different realizations are run to obtain a statistically meaningful result. This results in about 8000 avalanche events for each shearing rate. Figure 3 shows the event size distribution for different shearing rates. We find that the distribution follows a power law for small event sizes and cuts off at larger sizes, with the cutoff size depending on the shearing rate. The data are somewhat noisier towards the end of large event size because large events are rare.

Scaling behavior of the avalanches.—Analogous to the scaling behavior in models of crackling noise [14], we propose that there is a nonequilibrium critical point $v_0 = v_c$ in the system, and we expect, as $E \rightarrow \infty$, the data around the critical point to collapse in the form

$$P(E, \bar{v}) \sim E^{-\tau} f(E \bar{v}^\mu), \quad (6)$$

where $P(E, \bar{v})$ is the probability distribution of event energy E and $\bar{v} \equiv 1 - v_0/v_c$ is the reduced shearing rate, with v_0 being the shearing rate and v_c being the critical shearing rate. τ and μ are two critical exponents. As $\bar{v} \rightarrow 0$, $P(E, \bar{v})$ tends to a power law $P(E, \bar{v}) \sim E^{-\tau}$. Figure 4 shows an attempt to collapse the data, by using the equivalent scaling form $P(E, \bar{v}) \sim \bar{v}^{\tau\mu} g(E \bar{v}^\mu)$, with $\tau = 1.5$, $\mu = 2$, and $v_c = 1.5$ and universal scaling function $g(x) = x^{-\tau} f(x)$ shown in the collapse. Logarithmic binning is performed and singletons are ignored to obtain $P(E, \bar{v})$. We obtain a satisfactory data collapse over 4 decades, with

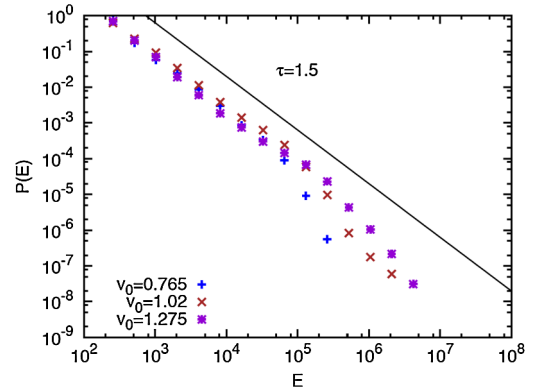


FIG. 3 (color online). The probability distribution of the event energy during dislocation avalanches, for different values of the shearing rates.

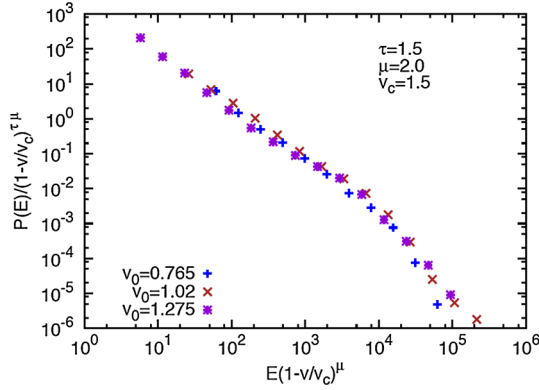


FIG. 4 (color online). Data collapse of the probability distribution of the event energy during dislocation avalanches, with $\tau = 1.5$, $\mu = 2$, and $v_c = 1.5$.

\bar{v} ranging from 0.15 to 0.49. As $E \rightarrow 0$, the collapse function $g(x)$ approaches the power law $g(x) \sim x^{-3/2}$, which agrees with $f(x) \rightarrow \text{const}$ in Eq. (6) for $x \rightarrow 0$ and $\tau = 3/2$. The collapse constrains the numerical exponent to the range $\tau = 1.5 \pm 0.2$. This agrees with the experimental result of $\tau = 1.5$ [28]. Similarly, for adiabatic stress increase in the pinned regime, Zaiser finds $\tau = 1.4$ [6]. Dimiduk *et al.* report $\tau = 1.5$ – 1.6 in experiments at fixed compression stress that leads to shearing [29]. In contrast to the universal distribution of time-integrated avalanche signals discussed here [see Eq. (5)], the signal amplitude statistics likely depend on details of the system: The distribution of energy amplitudes decays with the exponent $\tau = 1.8$ [9] in simulations and with $\tau = 1.6$ in experiments [9]. The acoustic emission intensity exponent is $\tau = 1.8$ in simulations at fixed stress [30].

Studies at adiabatically slow shear rate have suggested analogies to the depinning transition of magnetic domain walls [6,17,31], although one exponent that may deviate in simulations is discussed in Ref. [32]. The exponents found in our collapse are consistent with the domain wall depinning picture. As argued previously, mean field theory (MFT) is expected to give exact scaling results in this case. The MFT values for the exponent τ is $\tau = 1.5$ [6,17]. The MFT value for the exponent μ can be calculated from the MFT of Refs. [16,17,31] to be $\mu = 2$. These MFT values for the exponents τ and μ lead to a satisfactory collapse of the numerical avalanche size distributions at different shear rates. At zero temperature the critical shear rate is $v_c = 0$. Our simulations, however, are performed at finite temperature T . Temperature-induced dislocation creep causes the critical shear rate to appear to be nonzero, with the apparent $v_c(T) \rightarrow 0$ as $T \rightarrow 0$. It also causes the scaling collapse in Fig. 4 to be less precise than in zero temperature studies. We hope to report on a theoretical investigation of the temperature dependence and a comparison with experiments at finite shear rate in a future publication.

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