Carrier mobility with EPW

Tutorial Fri.3

Hands-on session

Hands-on based on Quantum Espresso 6.5

Linearised Boltzmann transport equation (BTE)

In this tutorial, we will show how to compute the intrinsic electron and hole mobility of hexagonal wurtzite GaN using the linearised iterative Boltzmann transport equation (IBTE) and the self-energy relaxation time approximation (SERTA).

The theory related to this tutorial can be found in the review Reports on Progress in Physics 83, 036501 (2020).

In particular the IBTE mobility refers to Eqs. 40, 41, 42 and 43 from the review while the SERTA mobility refers to Eqs. 41, 51 and 53. Note that an alternative formulation of the IBTE mobility is also given by Eqs. 63 and 64 by removing the magnetic field (B=0 limit).

Preliminary calculations with Quantum Espresso

First download the exercise files:

```
$ git clone https://github.com/wannier-developers/wannier-tutorials.git
$ cd 2020_03_0xford/3_epw/tuto_epw_mobility/
```

We first compute the dynamical matrix and perturbed potential (dVscf) using density functional perturbation theory on the irreducible Brillouin-Zone.

▶ To do so, compute the ground-state electron density using 2 cores:

```
scf.in
&control
    calculation='scf'
   prefix='gan'
    restart_mode='from_scratch'
    pseudo_dir='./'
    outdir='./'
    tprnfor=.true.
    tstress=.true.
    wf collect = .true.
    verbosity = 'high'
&svstem
    ibrav
             = 4
    celldm(1) = 5.9612
    celldm(3) = 1.6299
             = 2
    ntvp
    ecutwfc = 40
&electrons
    diagonalization='david'
    mixing_beta=0.7
    conv_thr=1.0d-14
    diago_full_acc = .true.
ATOMIC_SPECIES
 Ga 69.723 Ga-LDA.upf
N 14.007 N-LDA.upf
ATOMIC_POSITIONS crystal
 Ga 0.66666666667 0.33333333333 0.0
```

```
N 0.66666666667 0.333333333333 0.376458

Ga 0.33333333333 0.66666666667 0.5

N 0.33333333333 0.666666666667 0.876458

K_POINTS automatic
2 2 2 0 0 0
```

Note: For polar materials as is the case of GaN, the dielectric constant converges very slowly with the k-point grid (linearly). As a result, the k-point grid should be very dense (e.g. 20x20x20). In this example we use 2x2x2 for fast calculation.

- \$ mpirun -np 2 pw.x < scf.in | tee scf.out</pre>
- ► Compute the vibrational properties of GaN on a coarse 2x2x2 q-point grid.

```
--
&inputph

tr2_ph=1.0d-16,
prefix='gan',
amass(1)=69.723,
amass(2)=14.007,
outdir='./',
fildyn='gan.dyn',
fildvscf='dvscf'
ldisp=.true.,
epsil=.true.,
nq1=2, nq2=2, nq3=2
/
```

\$ mpirun -np 2 ph.x < ph.in > ph.out &

During the run, try looking in the ouput and locate the dielectric constants. You should get 23.12 in-plane and 17.23 for the out of plane direction. Those values are strongly overestimated due to the coarse k-point grid. The calculation should take about 5 min.

Note: The input variable responsible to produce the electron-phonon matrix element is fildvscf. Always make sure that this variable is present.

Note 2: Notice the very tight tr2_ph threshold parameter on the self-consistent first-order perturbed wavefunction. This is crucial to obtain good vibrational properties.

Finally, we need to post-process some of the data to make it ready for EPW. To do so, we can use a python script.

▶ Run the following python script which is inside the folder.

\$ python pp.py

The script will ask you to enter the prefix used for the calculation. In this case enter "gan". The script will create a new folder called "save" that contains the dvscf potential files and dynamical matrices on the IBZ.

Note: The script can usually be found in qe/EPW/bin/pp.py

The script should be compatible with python 2 and python 3 but if you are using python 2, the script might ask you to install the "future" library. If so, type:

\$ pip install --user future

For information, the converged phonon bandstructure of GaN should look like this:

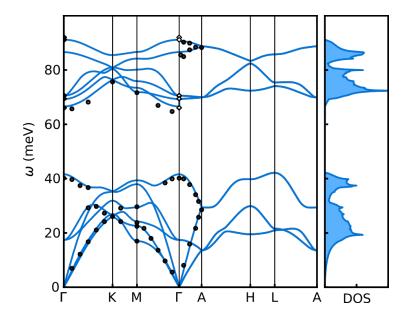


Figure 1: Phonon dispersion relations and phonon density of states of wurtzite GaN at the relaxed atomic positions from Phys. Rev. B **100**, 085204 (2019).

Interpolation of the electron-phonon matrix element in real-space with EPW

Because the earlier phonon calculation modified the density file, we have to do another DFT self-consistent calculation

```
$ mpirun -np 2 pw.x < scf.in | tee scf.out</pre>
```

▶ Do a non self-consistent calculation on a 2x2x2 uniform and Γ -centered k-point grid with crystal coordinates in the interval [0,1[

Such grid can be for example generated using python with

```
import numpy as np
N = 2
for ii in np.arange(0,1.0,1.0/N):
   for jj in np.arange(0,1.0,1.0/N):
     for kk in np.arange(0,1.0,1.0/N):
        print ii,' ',jj, ' ',kk, ' ',1.0/(N**3)
```

or using kmesh.pl as explained in the first tutorial.

But in any case the grid is included in the nscf.in file for you.

```
&control
    calculation='nscf'
    prefix='gan'
    restart_mode='from_scratch'
    pseudo_dir='./'
    outdir='./'
    tprnfor=.true.
    tstress=.true.
    wf_collect = .true.
    verbosity = 'high'

/
&system
    ibrav = 4
    celldm(1)= 5.9612
    celldm(3)= 1.6299
    nat = 4
```

```
ntyp
   ecutwfc = 40
   nbnd
            = 30
&electrons
   diagonalization='david'
   mixing_beta=0.7
   conv thr=1.0d-10
   diago_full_acc = .true.
ATOMIC SPECIES
 Ga 69.723
             Ga-LDA.upf
 N
     14.007
             N-LDA.upf
ATOMIC_POSITIONS crystal
 Ga 0.66666666667 0.33333333333 0.0
    0.66666666667 0.33333333333 0.376458
 Ga 0.333333333333 0.666666666667
                                  0.5
    0.33333333333 0.66666666667 0.876458
K_POINTS crystal
0.0
     0.0
           0.0
                0.125
0.0 0.0 0.5
               0.125
0.0
    0.5
           0.0
                0.125
    0.5
0.0
           0.5
                0.125
0.5
    0.0
           0.0
                0.125
0.5
     0.0
           0.5
                0.125
0.5
     0.5
           0.0
                0.125
0.5
    0.5
           0.5
                0.125
```

```
$ mpirun -np 2 pw.x -npool 2 < nscf.in > nscf.out
```

The reason for the non-self consistent calculation is that EPW needs the wavefunctions on the full BZ on a grid between 0 and 1.

Note: Since we are also interested in electron mobility, we will need the conduction bands. Notice that we added the input nbnd = 30 in nscf.in

▶ Perform an EPW calculation to Fourier-transform the electron-phonon matrix element from a coarse 2x2x2 \mathbf{k} and 2x2x2 \mathbf{q} -point grids to real space and then interpolate the electronic and phononic bandstructure along the $M-\Gamma-A$ high symmetry line.

```
$ mpirun -np 2 epw.x -npool 2 < epw1.in > epw1.out &
```

Note: The number of pool -npool has to be the same as the total number of core -np.

Look at the epw1.in input file. The following EPW input variables are important in our case:

```
vme
                          ! Compute exact velocities using Wannier interpolation
            = .true.
lpolar
            = .true.
                          ! This is a polar material
lifc
            = .false.
                          ! Do not use external IFC
            = 'simple'
                          ! Use the simple acoustic sum rule
asr tvp
use_ws
            = .true.
                          ! Use the optimal Wigner-Seitz cell
nbndsub
            = 14
                          ! We Wannierize only 14 bands
            = 12
nbndskip
                          ! We skip the first 12 bands
            = './save/'
{\tt dvscf\_dir}
                          ! Directory where the save folder generated by the \mathtt{@pp.py@} script is stored
                          ! We want to plot the electronic and/or phonon bandstructure
band_plot
            = .true.
                              (i.e. we do not use homogeneous fine grids)
            = './MGA.txt' ! K-point path from M to Gamma to A
filkf
            = './MGA.txt' ! Q-point path from M to Gamma to A
filaf
efermi_read = .true
                          ! Because our fine grids are non homogeneous,
                              the code cannot determine the Fermi level
fermi_energy= 11.80
                          ! We need to provide the Fermi level manually.
fsthick
            = 100.0
                          ! States that we want to take around the Fermi level.
                          ! We want all states so we put a large value.
```

Note: Because our k and q coarse grids are so coarse, we need use_ws. However it makes the calculation heavier (more vectors). If your coarse grids are large enough (6x6x6 or more), you probably can use Gamma centered WS cell and use use_ws = .false.

Note 2: You can provide the k/q-point path either in crystal or Cartesian coordinate. To do that, modify the second element of the first line of the MGA.txt file.

To have an idea of which Fermi level you should use, you can first do a calculation with a homogeneous fine grids by removing filkf and filqf and placing:

```
\begin{array}{lll} nk1 & = 12 \\ nk2 & = 12 \\ nk3 & = 12 \\ nq1 & = 12 \\ nq2 & = 12 \\ nq3 & = 12 \\ \end{array}
```

Finally, notice that at the end of epw1.in you need to provide the IBZ q-point list:

```
4 cartesian

0.000000000 0.000000000 0.000000000

0.000000000 0.00000000 -0.306767286

0.000000000 -0.577350269 0.00000000

0.000000000 -0.577350269 -0.306767286
```

This list has to be exactly the same as the one reported in the output of the phonon calculation at the beginning of ph.out that we did earlier.

The calculation should take about 1 min to run. At the end, the code should have produce two files band.eig and phband.freq which contain the interpolated electronic and phonon bandstructure along the M- Γ -A path. You can post-process those file yourself or used the plotband tool from Quantum Espresso by typing:

\$ plotband.x

Then follow the instruction. Read the plotband.x documentation if you have an issue. This will create a file which is graphable with gnuplot and should give you the following:

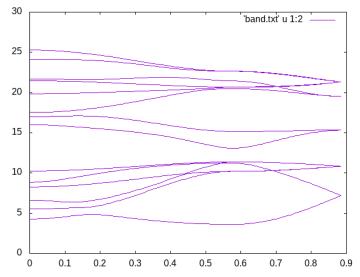


Figure 2: Wannierized electron bandstructure along the M-Γ-A path using EPW.

and the phonon bandstructure:

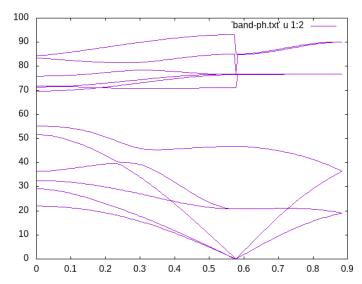


Figure 3: Wannierized phonon bandstructure along the M- Γ -A path using EPW.

Given that we have been using a 2x2x2 **k**-point and **q**-point grid and a low energy cut-off, such bandstructure quality is remarkable. Comparing with Figure 1, one can see that the only quantitative issue is related to the highest phonon-mode along the Γ -A direction, which improves with more converged results.

The electronic bandstructure can be compared to the following converged ones:

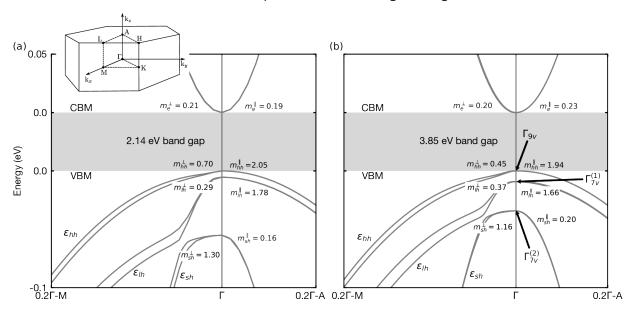


Figure 4: Electronic bandstructure of wurtzite GaN using (a) the LDA functional in the optimized ground-state LDA structure, and (b) quasiparticle G_0W_0 calculation. We indicate the effective masses at the zone center, obtained from the second derivatives of the band energy with respect to the wavevector along the ΓM and ΓA directions, respectively. The bandgap is off scale for clarity. We indicate the naming convention for the three topmost eigenstates at Γ . The energy levels have been aligned to the band edges. A schematic of the Brillouin zone of wurtzite GaN is given in the upper left corner. Figure from Phys. Rev. B 100, 085204 (2019).

The main difference with the calculation that we just did is that we did not include SOC in this tutorial due to time constrain. You perform a SOC calculation by adding

noncolin	= .true.,	
lspinorb	= .true.	

in the scf.in and nscf.in files inside the &system block.

Calculating the intrinsic drift mobility tensor of GaN as a function of temperature with EPW

We are now ready to compute the electron and hole mobility tensor for various temperatures. For this, run the following command to compute the hole mobility:

```
$ mpirun -np 4 epw.x -npool 4 < epw2.in | tee epw2.out</pre>
```

In this case, the code does a restart by reading the electron-phonon matrix element in real-space from the file gan.epmatwp1 and interpolate the matrix element on a fine k/q-point grid.

Note: For a restart, you can use an arbitrary number of cores (.e.g. we used 4 instead of 2). The maximum number of cores one can use corresponds to the maximum number of \mathbf{k} -points on the fine grid.

Before looking at the output, let us first discuss the specific input variables to perform transport calculations:

```
= .true. ! Required for a restart
kmaps
epwread
              = .true.
                        ! Tells the code to read the gan.epmatwp1 file
wannierize
              = .false. ! The Wannierization cannot be redone for a restart
             = .true. ! We wish to compute scattering rates
scattering
iterative_bte = .true. ! We wish to do IBTE and not just SERTA
mob_maxiter = 200
                        ! Maximum number of iterations for the IBTE
broyden_beta = 1.0
                        ! Linear mixing between iterations
int_mob
              = .false. ! We do not want to do electron and hole at the same time
carrier
              = .true. ! We want to specify carrier concentration
              = -1E13 ! We want 1E13 carrier per cm^3. The negative sign means hole
= .true. ! Use k-point symmetry to reduce the calculation cost
ncarrier
             = -1E13
mp_mesh_k
restart
              = .true. ! We want to create restart point during the interpolation
restart_freq = 50
                         ! A restart point is created every 50 q-points
              = .false. ! If .true. allows for reading the q-points within the fsthick
selecaread
              = .false. ! Once the transition probabilities have been written to file,
epmatkgread
                            allow for a restart to compute the mobilities
nstemp
              = 2
                        ! We are considering 2 temperatures.
              = 100
                        ! Initial temperature
tempsmin
tempsmax
              = 300
                        ! Final temperature
              = 0.0
                         ! Broadening of the Dirac deltas in the BTE.
degaussw
                        ! A value of 0 means adaptative broadening.
```

Note: There is a starting global and unique Fermi level given by fermi_energy or computed by the code. This Fermi level is used for the initialization of quantities and energy window determination. However the real Fermi level used for the transport calculation is determined automatically by the code depending on the required carrier concentration and temperature. There can even be up to two such real Fermi levels (i.e., one for electrons and one for holes).

We can take a look at the output of this second EPW calculation. We see the following:

```
Using uniform q-mesh: 12 12 12
Size of q point mesh for interpolation: 1728
Using uniform MP k-mesh: 12 12 12
Size of k point mesh for interpolation: 266
Max number of k points per pool: 68
```

which tells us that we have a fine \mathbf{q} -point grid with 1728 \mathbf{q} -points (no symmetry) and a fine \mathbf{k} -point grid with 266 \mathbf{k} -points (IBZ). Since we are doing a parallelization on \mathbf{k} -points, we will have 68 \mathbf{k} -points per core.

Next, the code will try to reduce the number of **q**-points we need to compute by taking the ones that contribute to the energy window. A file called selecq.fmt will be created and can be read later to avoid recomputing.

```
Number selected, total 50 148
Number selected, total 100 1443
Number selected, total 150 1715
We only need to compute 162 q-points
```

The code then reports the VBM and CBM as well as the Fermi level for the hole calculation for each temperature:

```
Valence band maximum = 11.381872 eV
Conduction band minimum = 13.076046 eV

Temperature 100.000 K
Mobility VB Fermi level = 11.517589 eV

Temperature 300.000 K
Mobility VB Fermi level = 11.804356 eV
```

The code also reports which states are included given the energy window:

```
Fermi Surface thickness = 0.400000 eV
This is computed with respect to the fine Fermi level 11.481872 eV
Only states between 11.081872 eV and 11.881872 eV will be included
```

The code then computes the transition probabilities but only saves the ones that are large enough:

```
Save matrix elements larger than threshold: 0.372108862978E-23
```

Note that the threshold is computed automatically based on the size of the system and is very conservative. The total error made by neglecting those matrix elements is lower than the machine precision.

Because we are using adaptive smearing, the code reports the minimum and maximum value of the broadening for that \mathbf{q} -point. This information is printed for every $\mathtt{restart_step}$.

```
Progression iq (fine) =
                             50/
 Adaptative smearing = Min:
                             1.414214 meV
                     Max:
                            293.598353 meV
Progression iq (fine) =
                            100/
                                      162
 Adaptative smearing = Min: 28.908200 meV
                     Max: 319.312222 meV
Progression iq (fine) =
                             150/
                                       162
  Adaptative smearing = Min:
                             48.994246 meV
                             149.060946 meV
                      Max:
```

At the end all the transition probabilities related to the valence bands are stored in the file gan.epmatkq1 as well as the support files sparsei, sparsej, sparsek, sparseq and sparset containing the band indexes, **k**-points, **q**-points and temperatures of the store transition probabilities. Similarly, files related to the conduction bands have the same name with the suffix "cb". Note that the file size is 28 Kb, orders of magnitude smaller than without the matrix element threshold. Finally, the code also produces two formatted text files called IBTEvel_sup.fmt and inv_tau.fmt containing the electronic velocities and scattering rates, respectively.

Note: The interpolation and storing of the transition probabilities to file can take a lot of time. If the calculation crashes during runtime (due to time-out for example), just put selecqread == .true. to avoid recomputing the selecq.fmt and relaunch. The code will automatically continue.

Once all the transition probabilities are computed, you can always restart the calculation by using the epmatkgread == .true..

Note that here the code did that for you automatically:

```
epmatkqread automatically changed to .TRUE. as all scattering have been computed.
```

Then the code computes the SERTA mobility:

emp [K] ======	Fermi [eV]	Hole density [cm^-3]	Population SR [h per cell]		Hole mobility [cm^2/Vs]	
0.000	11.5176	0.10000E+14	0.12925E-25	0.767956E-02	0.000000E+00	0.000000E+00
				0.000000E+00 0.000000E+00	0.767956E-02 0.000000E+00	0.000000E+00 0.573540E+04
0.000	11.8044	0.10000E+14	0.00000E+00	0.585711E+01 0.000000E+00 0.000000E+00	0.000000E+00 0.585711E+01 0.000000E+00	0.000000E+00 0.000000E+00 0.184230E+03

The GaN in-plane mobility (xx and yy components) is 0.767956E-02 and the out-of plane one (zz) is 0.573540E+04. Those values are totally not converged and physically wrong because of the very coarse fine grids that we used. Typical grids are larger than $100\times100\times100$. One would always expect the mobility at higher temperature to be lower than the one at lower temperature. This is correctly the case of the out-of plane component but not for the in-plane one.

Note: The population sum-rule is simply the sum of all change of population due to the external electric field which should be 0. If this value is large, there is probably something wrong in the calculation.

Finally, the code starts solving the BTE and converges after 36 iterations to:

Iteration	number:	36				
Temp [K]	Fermi [eV]	Hole density [cm^-3]	Population SR [h per cell]		Hole mobility [cm^2/Vs]	
100.000	11.5176	0.10000E+14	-0.12824E-25	0.896091E-02	0.000000E+00	0.000000E+00
				0.000000E+00	0.896091E-02	0.00000E+00
				0.000000E+00	0.00000E+00	0.374318E+04
300.000	11.8044	0.10000E+14	0.13235E-22	0.757728E+01	0.00000E+00	0.000000E+00
				0.000000E+00	0.757728E+01	0.00000E+00
				0.00000E+00	0.00000E+00	0.185868E+03
				0.806237E-06	Max error	

To conclude this exercise, we launch the last calculation with:

```
$ mpirun -np 2 epw.x -npool 2 < epw3.in | tee epw3.out</pre>
```

Note: You may need to remove the restart_ibte.fmt file since this is a different calculation and not the restart of the previous one.

In this case, we compute the electron mobility. Take some time to analyse the differences between epw3.in and epw2.in to understand how to do such calculations. It should be quite self-explanatory. The fine ${\bf k}$ and ${\bf q}$ point grids need to be much denser for real calculations. However, we can already get relatively decent results. Try filling the table below:

Temperature (K)	hole ε_F (eV)	hole μ (cm ² /Vs)	electron ε_F (eV)	electron μ (cm ² /Vs)
100				
200				
300				
400				
500				

For information, by using much denser grids as well as SOC and GW corrections on the eigenvalues, one can get a fair agreement with experiment as shown in the following figure:

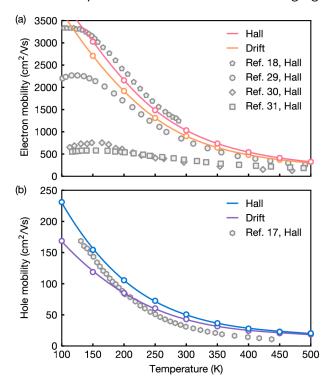


Figure 5: Electron (a) and hole (b) mobilities of wurtzite GaN, calculated using the IBTE, compared with the experimental data. We show both the drift mobilities and the Hall mobilities obtained by applying the Hall factor (not discussed here). Figure from Phys. Rev. Lett. **123**, 096602 (2019).

▶ Try to increase the fine grids and add a few more temperatures and see if you can get a result closer to convergence.

Suggestions for efficient transport calculations

Transport calculations are computationally heavy as they require a very fine sampling of both the \mathbf{k} -point and \mathbf{q} -point grids. However, various schemes have been implemented in EPW to reduce the computational cost.

- The memory scales linearly with the number of temperature nstemp values. Try decreasing it if you have memory issues. The cpu times scales strongly sub-linearly with the number of temperature values. Therefore try to do as many temperatures as your memory allows.
- Low temperatures span a smaller energy window and are therefore more difficult to converge. You can decrease fsthick to make the calculation more efficient. Note however that a single fsthick is used for all temperatures and therefore the fsthick needs to be large enough to accommodate the largest temperature.
- Using int_mob == .true. is convenient because it allows for the calculation of both electron and hole mobility at the same time. However, the code considers only the states within an energy window given by fsthick around the starting Fermi level given by fermi_energy. As a result you might need to use a larger than needed fsthick. The recommended way is therefore to compute electron and hole mobility in two separate runs.

• Transport properties only require states very close to the band edge and significant speedup can be achieved by using a small fsthick. Note that you need to converge your result with increasing values of fsthick and that larger values are required for higher temperature. For example in this tutorial we used a 0.4 eV fsthick but we placed our Fermi level 0.1 eV above the valence band maximum (VBM). As a result we are considering only the states up to 0.3 eV above the VBM. This is a typical order of magnitude.