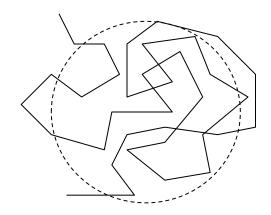
A polymer is a molecule comprised of many similar units. Unlike simple molecules polymers do not have a unique molecular weight but have a distribution. When dissolved in solution the molecules swell up to much larger dimensions.



There are two ways by which we can define the dimensions of a polymer chain, one is the root mean square end to end distance, $\langle r^2 \rangle^{1/2}$, or the root mean square distance of the elements of the chain from its centre of gravity, $\langle s^2 \rangle^{1/2}$, also known as the radius of gyration. These two parameters can be linked by the simple expression

$$\left\langle r^2 \right\rangle^{1/2} = \left\langle 6s^2 \right\rangle^{1/2} - - -3.1$$

 A simple model of a polymer chain consists of a series of x links of length, I, joined in a linear sequence, with no restrictions on the bond angle. Using this model the end to end distance is proportional to the square root of the number of links:

$$\langle r^2 \rangle^{1/2} = lx^{1/2} - -3.2$$

 Thus the size of a polymer molecule is proportional top the square root of its molecular weight.

- The radius of gyration is more often quoted experimentally because it can be measured experimentally, using light scattering, sedimentation, etc.
- Radius of gyration for PEO

Molecular weight	$< r^2 > 1/2$	$<$ S ² > $^{1/2}$
	nm	nm
100,000	12.1	4.9
200,000	16.7	6.8
1,000,000	38.0	15.5
5,000,000	85.6	35.0

Polymer Molecular Weight

- Polymers do not have a unique molecular weight, they have a distribution.
 The average of that distribution depends on how you calculate (measure) the molecular weight.
- Number average Molecular weight (osmotic pressure)

$$M_{N} = \frac{\sum_{i} n_{i} M_{i}}{\sum_{i} n_{i}}$$

Weight Average Molecular weight (light scattering)

$$M_{w} = \frac{\sum w_{i} M_{i}}{\sum w_{i}} = \frac{\sum n_{i} M_{i} M_{i}}{\sum n_{i} M_{i}} = \frac{\sum n_{i} M_{i}^{2}}{\sum n_{i} M_{i}}$$

n_i and w_i are the number and weight fractions respectively

 M_W is always > M_N and the ratio M_W/M_N is often taken as a measure of the molecular weight distribution

What do the Weights Mean?

Mn: This gives you the number average molecular weight

Let's say you had the following polymer sample:

2 chains:	1,000,000	2,000,000
5 chains:	700,000	3,500,000
10 chains:	400,000	4,000,000
4 chains:	100,000	400,000
2 chains:	50,000	100,000
		10,000,000

Mn = 435,000 10,000,000/23 = 435,000

Weight Average Molecular Weight

Mw: Since most of the polymer mass is in the heavier fractions, this gives the average molecular weight of the most abundant polymer fraction by mass.

$$\frac{2,000,000}{10,000,000} = 0.20 \times 1,000,000 = 200,000$$

$$\frac{3,500,000}{10,000,000} = 0.35 \times 700,000 = 245,000$$

$$\frac{4,000,000}{10,000,000} = 0.40 \times 400,000 = 160,000$$

$$\frac{400,000}{10,000,000} = 0.04 \times 100,000 = 4,000$$

$$\frac{100,000}{10,000,000} = 0.01 \times 50,000 = 500$$

$$M_{\rm w} = 609,500$$

Thermodynamics of Simple Liquid Mixtures

 From the condition for equilibrium between 2 phases may be derived the free energy for dilution of a solution,

$$\Delta G_A = kT \ln \left(\frac{p_A}{p_A^0} \right) - -3.3$$

 Where ΔG is the free energy of dilution, resulting from the transfer of one molecule of liquid A from the pure liquid state with vapour pressure p_A⁰ to a large amount of solution with vapour pressure, p_A.

Ideal Solutions

Assuming the molecules of component A and B have roughly the same size, shape and interactions. They may then form an ideal solution, defined as one in which Raoult's Law is obeyed, which states that the partial pressure of each component is equal to its mole fraction, *n*, thus

$$\Delta G_A = kT \ln n_A - -- 3.4$$

So that the total free energy of mixing is

$$\Delta G = N_A \Delta G_A + N_B \Delta G_B$$

$$\Delta G = kT(N_A \ln n_A + N_B \ln n_B) - --3.5$$

N is the number of molecules

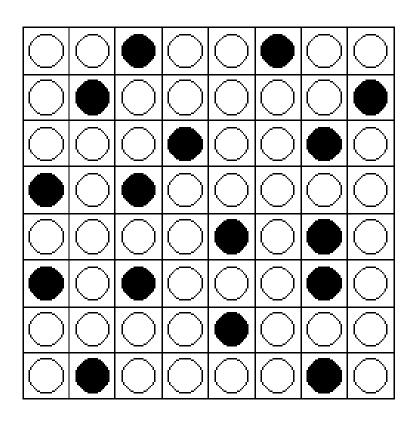
 The conditions for ideal mixing ΔH=0 so that the entropy of mixing is given by

$$\Delta S = -k(N_A \ln n_A + N_B \ln n_B) - --3.6$$

 Which is positive for all compositions, so that by the 2nd law spontaneous mixing occurs in all proportions

- Other types of Mixing
 In practice few systems obey
 Raoult's Law and deviations
 may occur.
- 1. Athermal Solutions where $\Delta H=0$ but ΔS no longer given by Eqn 3.6
- 2. Regular Solutions where ΔS has the ideal value, but ΔH is finite
- 3. Irregular Solutions where both ΔH and ΔS are non ideal.

A mixture of two liquids

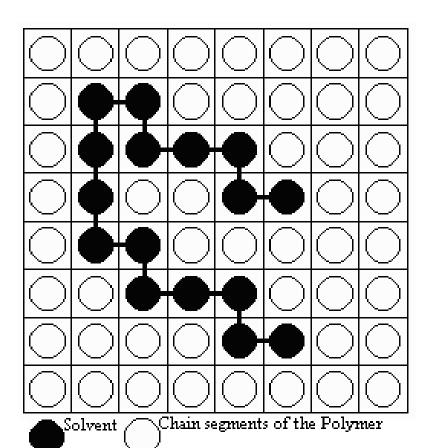


Let us construct a lattice with 64 lattice sites and put 48 molecules of solvent A and 16 molecules of solvent B.

The entropy is now given by

$$\Delta S = k(N_A \ln n_A + N_B \ln n_B) - --3.6$$

where $N_A = 48$, $N_B = 16$, $n_A = 0.75$, $n_B = 0.25$,



Now let us construct a lattice with 64 lattice sites and put 48 molecules of solvent A and 1 molecule of polymer B.

The entropy is now given by

$$\Delta S = k(N_A \ln \phi_A + N_B \ln \phi_B) - --3.6$$

where N_A =48, N_B =1, and ϕ_A the volume fraction of solvent =0.75, ϕ_B , the volume fraction of polymer=0.25.

Thus the entropy of the polymer is considerably less compared to the solution

Ideal solutions assume that all intermolecular interactions in a solution are the same, so that the enthalpy of mixing is zero. The simplest correction is to introduce a parameter (△ɛ) that is a measure of the difference between like-unlike and like-like molecular interactions, where

$$\Delta \varepsilon = \varepsilon_{A,B} - \frac{1}{2} (\varepsilon_{AA} + \varepsilon_{BB}) - --3.7$$

 This gives the Enthalpy of mixing of mixtures A and B as

$$\Delta H_{mix} = z N_A n_B \Delta \varepsilon - - - - - - 3.8$$

Defining

$$\chi = \frac{z\Delta\varepsilon}{kT} - - - -3.9$$

We obtain

$$\Delta H_{mix} = kTN_A n_B \chi - --3.10$$

- For polymer solutions we use the same arguments but replace the mole fraction of component B, by the volume fraction of polymer present, φ_B.
- i.e

$$\Delta H_{mix} = kTN_A \phi_B \chi - --3.11$$

 So that the total free energy of mixing is given by

$$\Delta G = kT(N_A \ln \phi_A + N_B \ln \phi_B + \chi N_A \phi_B) - - -3.12$$

 From equation 3.11 it can be seen that as long as χ is positive, which it virtually always is, the enthalpy of mixing is always positive, and is unfavourable. Thus one would expect phase separation to occur on cooling, since on cooling the entropic term is reduced. This is indeed what one observes in many systems. However, as we know from cooking eggs etc., this is not true. In fact for water soluble polymers this is not true at all.

- The reason being that in our analysis we have assumed that χ is independent of temperature. This is not true for water as a solvent due to H-bonding, which decreases as the temperature increases.
- From equation 3.12 we can estimate the change in chemical potential on mixing a polymer

$$\mu_A - \mu^0 = \left(\frac{d\Delta G}{dN_1}\right)_{N_B}$$

$$= kT \left[\ln \phi_A + \left(1 - \frac{1}{x} \right) \phi_B + \chi \phi_B^2 \right] - - 3.13$$

Where x is the number of segments of polymer. If the logarithmic term in eqn. 3.13 is expanded and only low powers of the polymer volume fraction φ_B are retained we end up with a virial expansion

$$\mu_A - \mu^0 = kT \left[\frac{\phi_B}{x} + \left(\frac{1}{2} - \chi \right) \phi_B^2 + \dots \right] - \dots - 3.14$$

 Or in terms of osmotic pressure, ∏

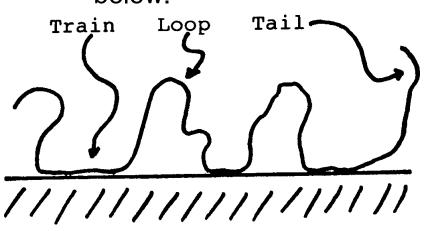
$$\Pi = \frac{kT}{V_A} \left[\frac{\phi_B}{x} + \left(\frac{1}{2} - \chi \right) \phi_B^2 + \dots \right] - \dots - 3.15$$

• This is very analogous to the virial equation for gases. The 1^{st} term is the ideal term and the 2^{nd} a non-ideal term. It can be seen that when $\chi = 0.5$ the properties of polymer solution are ideal. This is known as the θ condition.

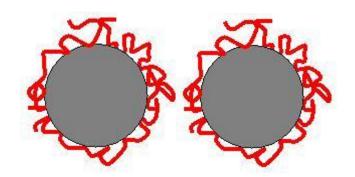
 It turns out that when χ< 0.5, the polymer dissolves the solvent is known as a good solvent, but when χ> 0.5, the polymer does not dissolve and the solvent is a poor solvent.

Polymer adsorption

 When polymers adsorb to a surface they generally do not lay flat on the surface, but extend away from the surface as well. The polymer segments thus exist as a series of tails, loops and trains as shown below.



This means that when
 polymers are adsorbed onto
 particle surfaces, they may
 interfere with each other before
 van der Waals forces play a
 significant role and aggregate
 the particles



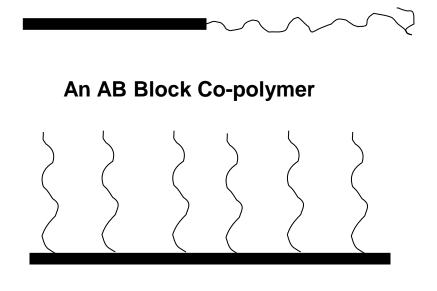
Polymer adsorption

In the above case the adsorbed polymer provides a physical buffer, which prevents the cores of the particles from coming together and hence prevents the van der Waals forces being significant. A polymer when it is adsorbed to a surface generally does not form a compact layer, but a fuzzy, relatively thick monolayer id formed, a bit likje a ball of cotton wool. The polymer layer itself may only contain around 5% polymer, the rest is solvent.

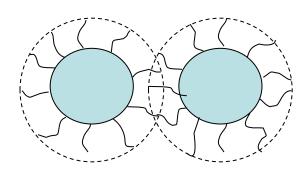
The criteria for polymer to be a good stabilising agent is for it to be strongly adsorbed onto the surface and to extend sufficiently far away from the surface (5-10nm) for van der Waals forces to be less than kT. These conditions are mutually exclusive for homopolymers, i.e. a polymer that is made of only one component. If the polymer adsorbs too strongly it will lay flat on the surface and the van der Waals forces will be significant, whilst the other limiting condition is such that the polymer will not adsorb at all

Polymer adsorption

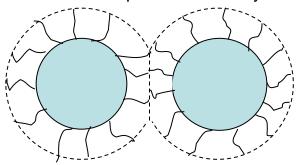
 If the polymer is comprised of two parts, where on part of the polymer will adsorb onto the surface and the other not, then a good stabiliser will be prepared. Such materials are called copolymers and block and graft copolymers are materials that fulfil this role well. They are in effect similar to surfactants.



A Graft or Comb copolymer



Overlap of adsorbed layer



Compression of adsorbed layer

When two particles coated with polymers interact the two polymer layers overlap and compress. This has two effects one osmotic, enthalpic, and one entropic. The osmotic effect is that the concentration of polymer in the gap is increased, this increases the osmotic pressure and the particles are force apart. The entropic effect is a consequence of the compression of the polymer layer decreasing the number of configurations the polymer can adopt. Both effects have been quantified but the theory is not as good as that of double layers or van der Waals interactions

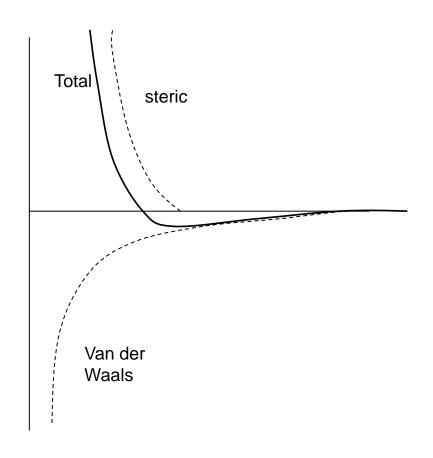
 In the simplest approach, it is assumed that the total interaction is simply the sum of the osmotic and entropic components.

$$\Delta G_{mix} = \frac{2\pi RkTV_{p}^{2}n^{2}\Gamma^{2}(\frac{1}{2}-\chi)}{V_{I}}S_{os} --3.16$$

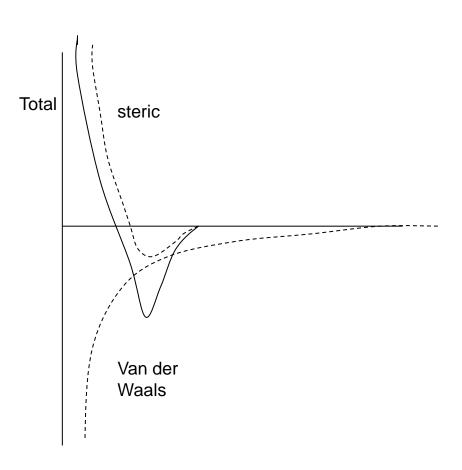
- Where V_p and V_l are the molar volume of the polymer segments and the liquid respectively, n is the number of polymer segments overlapping and Γ is the amount of polymer adsorbed, and S_{mix} is a geometric term that depends on the particle size and the thickness of the polymer layer.
- Inspection of eqn. 3.16 shows that the sign of the free energy of mixing depends on the value of χ , the Flory-Huggins interaction parameter. If χ > 0.5 then the mixing term is –ve and an attraction is seen, if χ < 0.5, then the mixing term is positive and we get a repulsive interaction, and if χ =0.5 then the mixing term is zero.
- The entropic term is always repulsive though since as the surfaces come closer together the number of configurations the polymer can adopt is reduced.

$$\Delta G_{ent} = 2\pi RkT\Gamma.S_{ent} - --3.17$$

- Where S_{ent} is another geometric term.
- Opposite is the total interaction curve, including van der Waals forces when the χ parameter <0.5. Note that the depth of the minimum may be adjusted by changing the thickness of the polymer layer (Molecular weight).



• Opposite is the total interaction curve, including van der Waals forces when the χ parameter >0.5



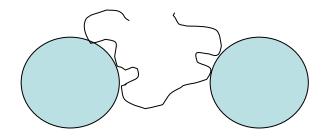
- We can see from these graphs that the type of interaction attraction or repulsion, is dependent on the value of the χ parameter. The χ parameter itself is dependent on temperature as we have seen, thus flocculation (aggregation) of particles is also dependant on temperature. Since χ decreases as T increases for most solvent soluble polymers the same is true regarding flocculation.
- For water soluble polymers the opposite is true

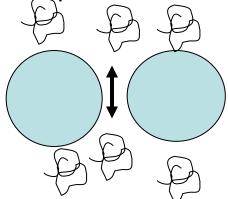
Stabilising Polymer	M.Wt	Dispersion Medium	Flocculation Temperature	Θ Temperature
Polyethyleneoxide	1 400	0.39M MgSO ₄	317 K	330 K
Polyethyleneoxide	4 700	0.39M MgSO ₄	320 K	327
Polyethyleneoxide	10 000	0.39M MgSO ₄	318 K	319
Polyethyleneoxide	13 500	0.39M MgSO ₄	318 K	315
Polyethyleneoxide	23 000	0.39M MgSO ₄	314 K	315
Polyacrylic acid	9 800	0.2 M HCI	287 K	287
Polyacrylic acid	19 300	0.2 M HCI	289	289
Polyacrylic acid	51 900	0.2 M HCI	283	287
Polyacrylic acid	89 700	0.2 M HCI	281	287
Polyvinylalcohol	26 000	2.0M NaCl	320 K	300
Polyvinylalcohol	57 000	2.0M NaCl	301	300
Polyvinylalcohol	270 000	2.0M NaCl	312	302
Polyacrylamide	18 000	2.1M (NH ₄) ₂ SO ₄	292 K	290
Polyacrylamide	60 000	2.1M (NH ₄) ₂ SO ₄	295	290
Polyacrylamide	180 000	2.1M (NH ₄) ₂ SO ₄	280	290

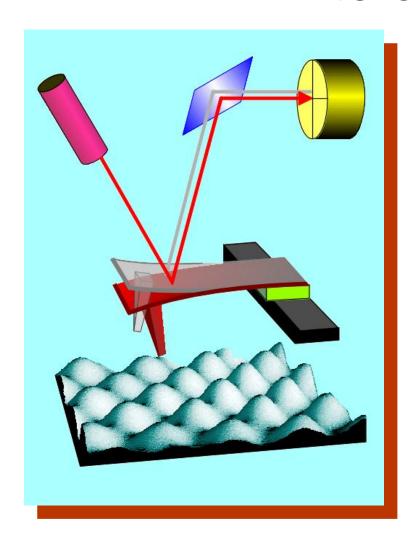
Flocculation of Particles by Polymer

Under some circumstances
 polymers can flocculate
 particles. Under very dilute
 polymer concentrations of
 polymer, so that the particles
 are not fully covered by
 polymer, aggregation occurs
 by one polymer molecule
 bridging between two particles

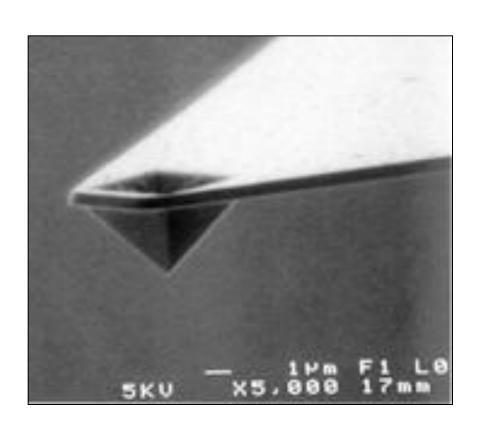
Also if the polymer is non adsorbing, when the gap between two particles becomes less than the dimension of the polymer, there will be less polymer between the two particle surfaces than in the bulk. Thus solvent will flow from the dilute to the concentrated region. This is brought about by the particles moving towards each other. This is known as depletion flocculation.

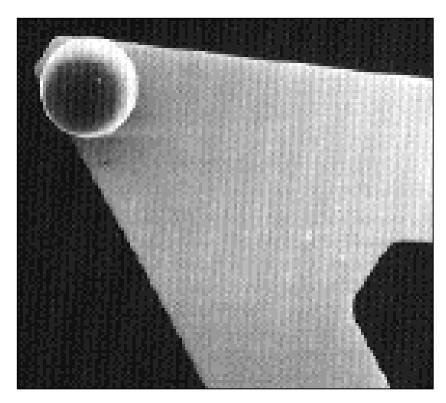




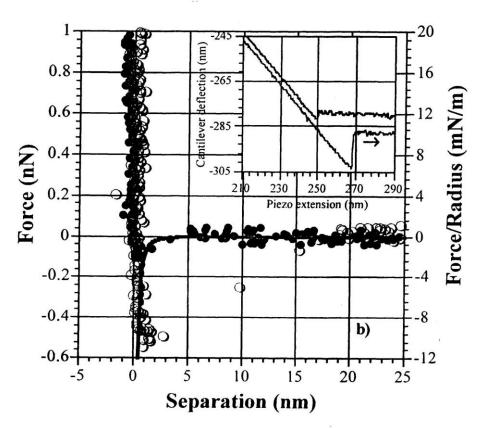


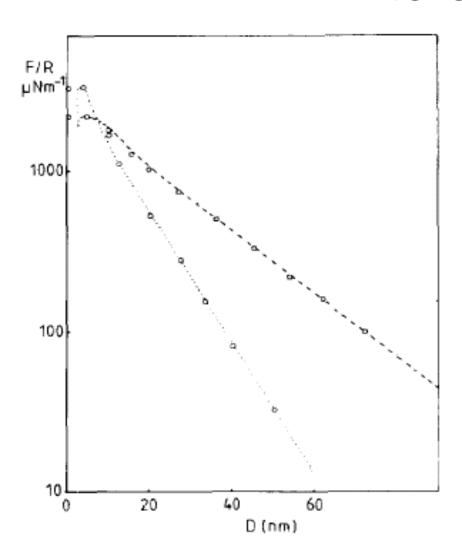
Particle interactions can be measured by modifying an atomic force microscope. In this experiment the interaction between a sharp tip attached to the end of a cantilever can be determined by monitoring the bending of the cantilever using laser light. By replacing the tip by a particle, particle surface and particle-particle interactions can be measured.



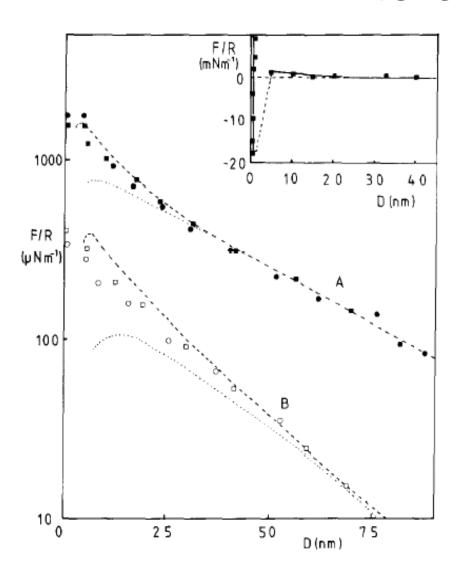


 Silicon Nitride Silicon Nitride in Bromonaphthalene





The force profile between two curved mica sheets immersed in (i) water (effective electrolyte solution of 1.1 x 10⁻⁵ mol dm⁻³ (\Box) and (ii) 1.0 x 10⁻³ mol dm-3 potassium nitrate solution (o) The results have been fitted assuming a surface potential of 110 mV for an interaction at constant surface potential (broken line for water and dotted line for 1.0 x 10-3 mol dm-3 potassium nitrate solution).



The interaction force profile between two curved mica sheets bearing poly-l-lysine adsorbed from water, molecular weight 4,000 g mol/l, at a solution concentration of 25 mg/l. curve A black symbols. Curve B shows the results obtained when the poly-1-lysine solution was replaced by water. The results for curve A have been fitted (for details see text) assuming a surface potential of 80 mV and an electrolyte concentration of 7.9x10⁻⁵ mol/l, for constant surface charge (broken line) and constant surface potential (dotted line) interactions. Curve B has been fitted assuming a surface potential of 30 mV and an electrolyte concentration of $1.9 \times 10^{-4} \text{ mol.l.}$

