Evaluation of the Virtual Crystal Approximation for Predicting Alloy Vibrational Mode Properties and Thermal Conductivity

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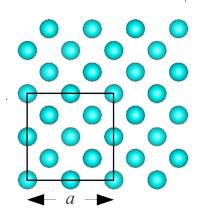
http://ntpl.me.cmu.edu/

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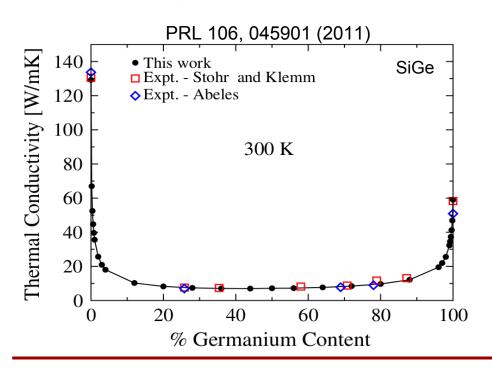


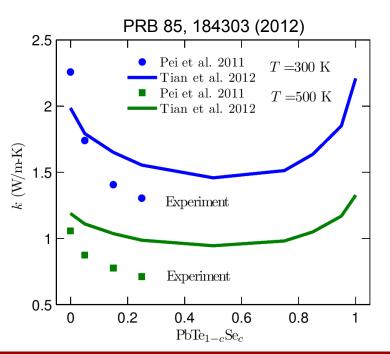
Motivation: experimental accuracy

Density Functional Theory (DFT) + Virtual Crystal (VC) approximation + Anharmonic Lattice Dynamics (ALD) (**VC-ALD**)

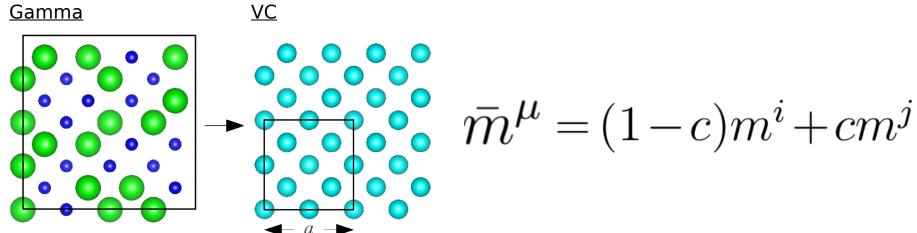


Alloys: isotopic effects, thermoelectric materials



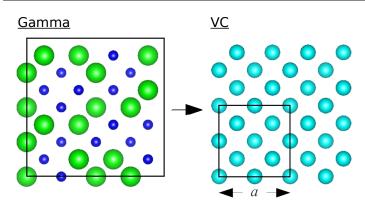


Virtual Crystal Approximation



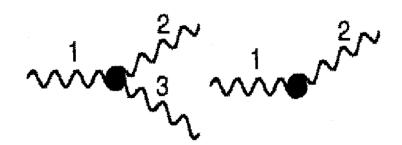
$$k_{ph,\mathbf{n}} = \sum_{\kappa} \sum_{\nu} \frac{k_B}{V} D_{ph,\mathbf{n}} \binom{\kappa}{\nu}$$
$$D_{ph,\mathbf{n}} \binom{\kappa}{\nu} = v_{g,\mathbf{n}}^2 \binom{\kappa}{\nu} \tau \binom{\kappa}{\nu}$$

VC-ALD Diffusivities

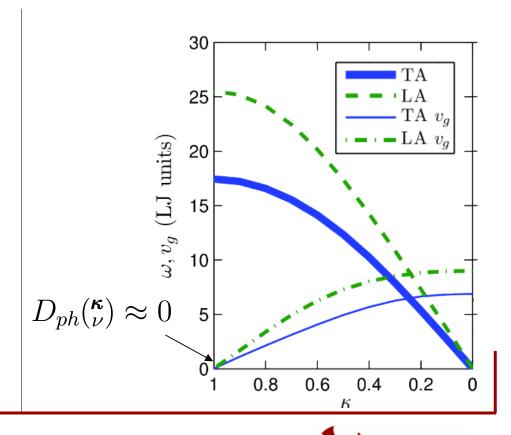


$$D_{ph,\mathbf{n}}(^{\kappa}_{\nu}) = v_{g,\mathbf{n}}^2(^{\kappa}_{\nu}) \, \tau(^{\kappa}_{\nu})$$

Matthiessen's Rule

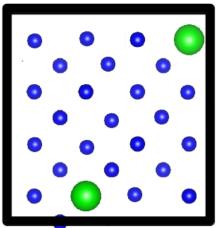


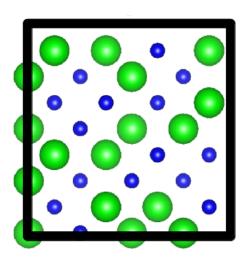
$$\frac{1}{\tau({}^{\kappa}_{\nu})} = \frac{1}{\tau_{p-p}({}^{\kappa}_{\nu})} + \frac{1}{\tau_{p-d}({}^{\kappa}_{\nu})}$$



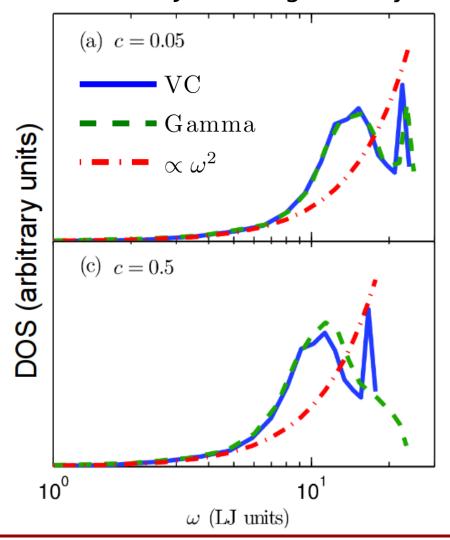
Explicit disorder: VC vs Gamma



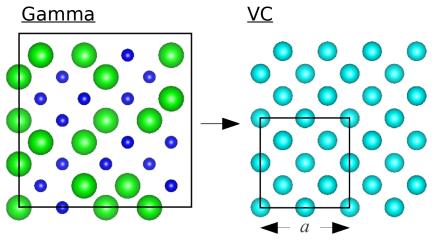




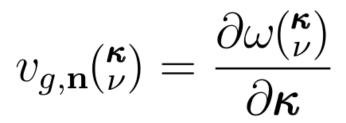
Lennard-Jones Argon Alloys

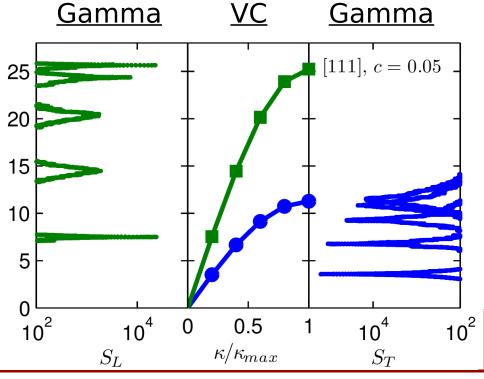


Explicit disorder: Structure Factor



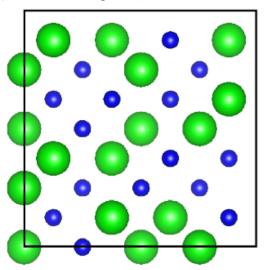
$$D_{ph,\mathbf{n}}({}^{\boldsymbol{\kappa}}_{\boldsymbol{\nu}}) = v_{g,\mathbf{n}}^2({}^{\boldsymbol{\kappa}}_{\boldsymbol{\nu}}) \, \tau({}^{\boldsymbol{\kappa}}_{\boldsymbol{\nu}})$$





Normal Mode Decomposition (NMD)

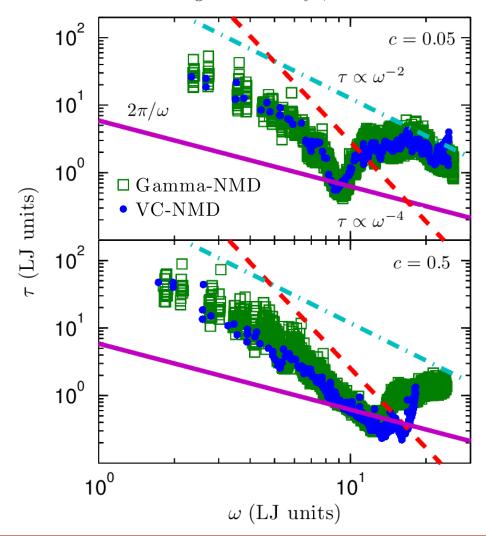
Molecular Dynamics Gamma



$$D_{ph,\mathbf{n}}({}^{\boldsymbol{\kappa}}_{\boldsymbol{\nu}}) = v_{g,\mathbf{n}}^2({}^{\boldsymbol{\kappa}}_{\boldsymbol{\nu}}) \tau({}^{\boldsymbol{\kappa}}_{\boldsymbol{\nu}})$$

$$\tau(^{\kappa}_{\nu}) = \int_0^{t^*} \frac{\langle E(^{\kappa}_{\nu}; t) E(^{\kappa}_{\nu}; 0) \rangle}{\langle E(^{\kappa}_{\nu}; 0) E(^{\kappa}_{\nu}; 0) \rangle} dt$$

LJ Argon and Alloys, T=10 K

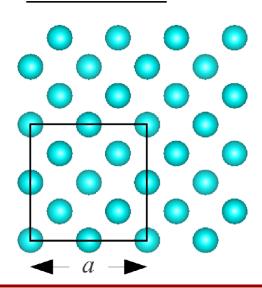


VC Diffusivities

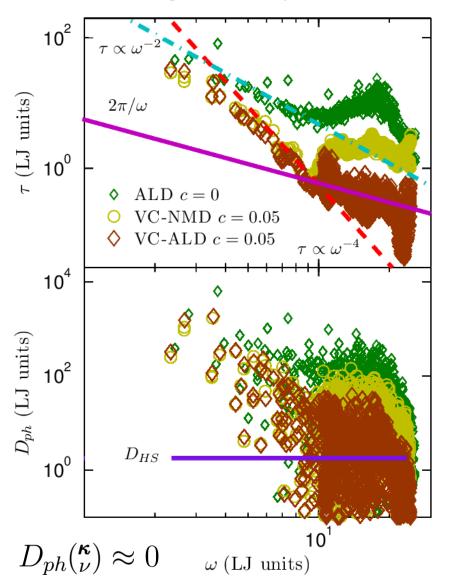
$$D_{ph,\mathbf{n}}(^{\kappa}_{\nu}) = v_{g,\mathbf{n}}^2(^{\kappa}_{\nu}) \, \tau(^{\kappa}_{\nu})$$

$$D_{HS} = \frac{1}{3}v_s a$$

VC Unit Cell

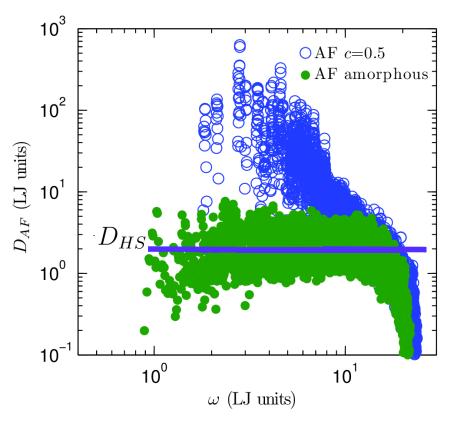


LJ Argon and Alloys, T = 10 K

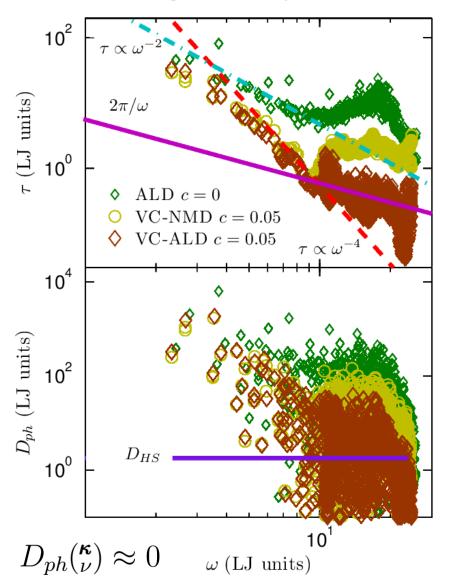


AF Diffusivities

$$k_{AF} = \sum_{diffusions} \frac{k_B}{V} D_{AF,i}(\omega_i)$$



LJ Argon and Alloys, T = 10 K



Thermal conductivity

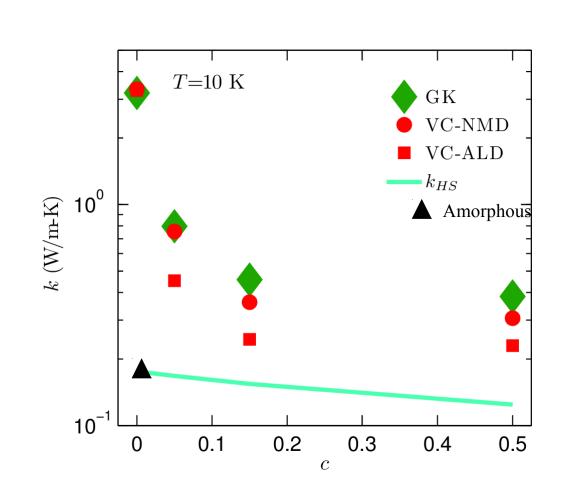
<u>Green-Kubo (GK): top-down</u> <u>method, no assumptions</u>

High-scatter adjustment*:

$$D_{ph}({}^{\kappa}_{\nu}) < D_{HS}$$

$$D_{ph}({}^{\kappa}_{\nu}) = D_{HS}$$

$$k_{HS} = \frac{k_B}{V_b} b v_s a$$



Thermal conductivity

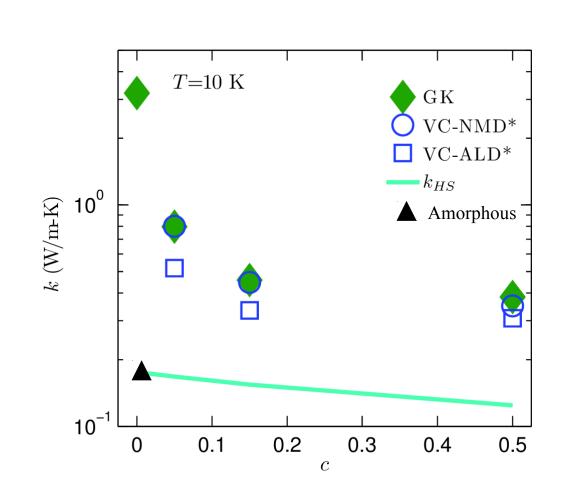
Green-Kubo (GK): top-down method, no assumptions

<u>High-scatter adjustment*:</u>

$$D_{ph}({}^{\kappa}_{\nu}) < D_{HS}$$

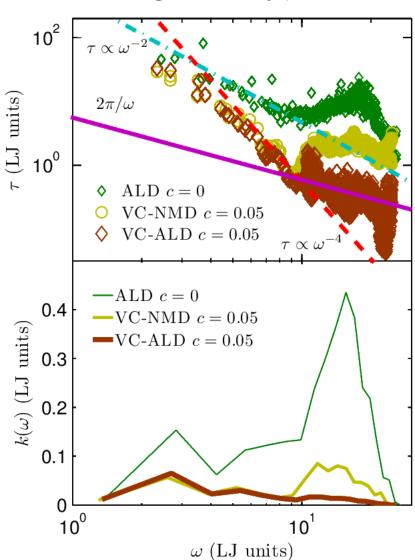
$$D_{ph}(^{\kappa}_{\nu}) = D_{HS}$$

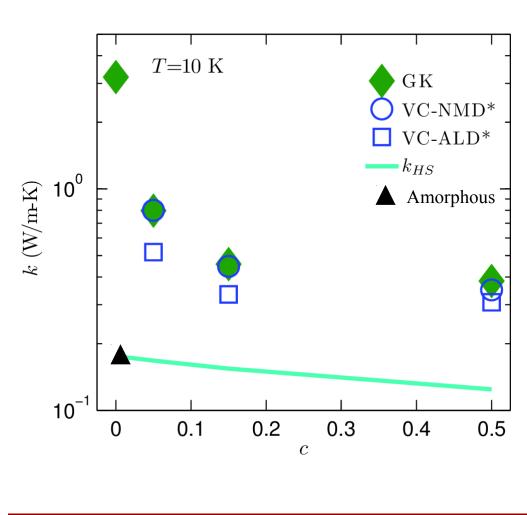
$$k_{HS} = \frac{k_B}{V_b} b v_s a$$



Thermal conductivity spectrum

LJ Argon and Alloys, T = 10 K





<u>Summary</u>

<u>VC approximation underpredicts</u> mode group velocities at high frequency, which lead to underprediction of mode diffusivity.

VC-ALD underpredicts lifetimes at high-frequency.

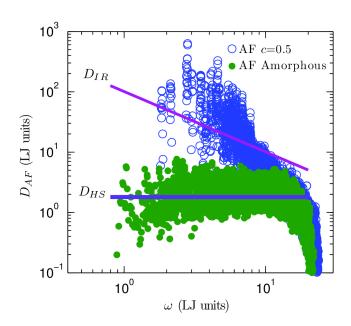
LJ Argon has important contribution from high-frequency modes to thermal conductivity.

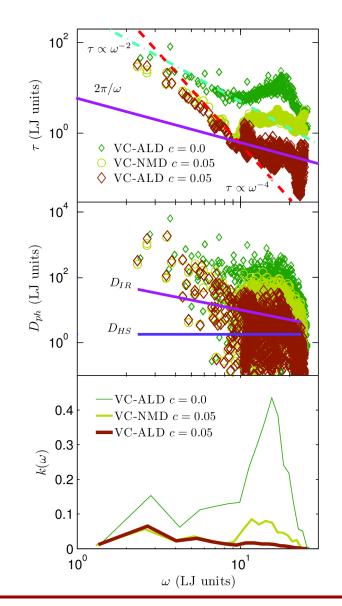
<u>Breakdown of VC-ALD method</u> is likely for materials with thermal conductivity near the high-scatter limit, or for modes below the high-scatter limit.

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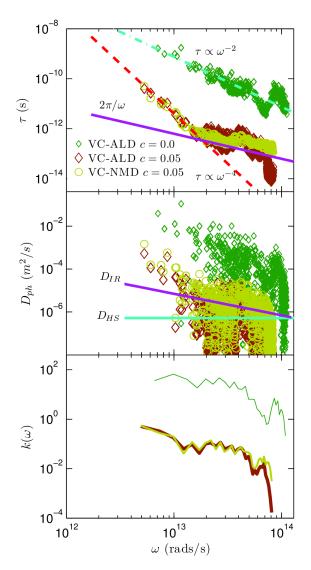
HS/IR Limit

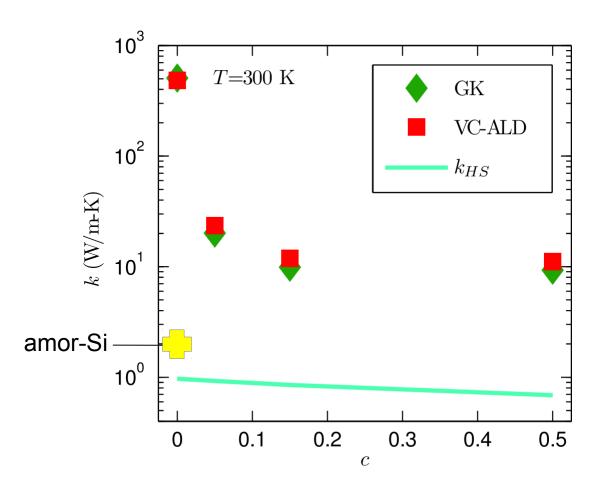
$$D_{IR} = \frac{2\pi}{3} \frac{v_s^2}{\omega}.$$





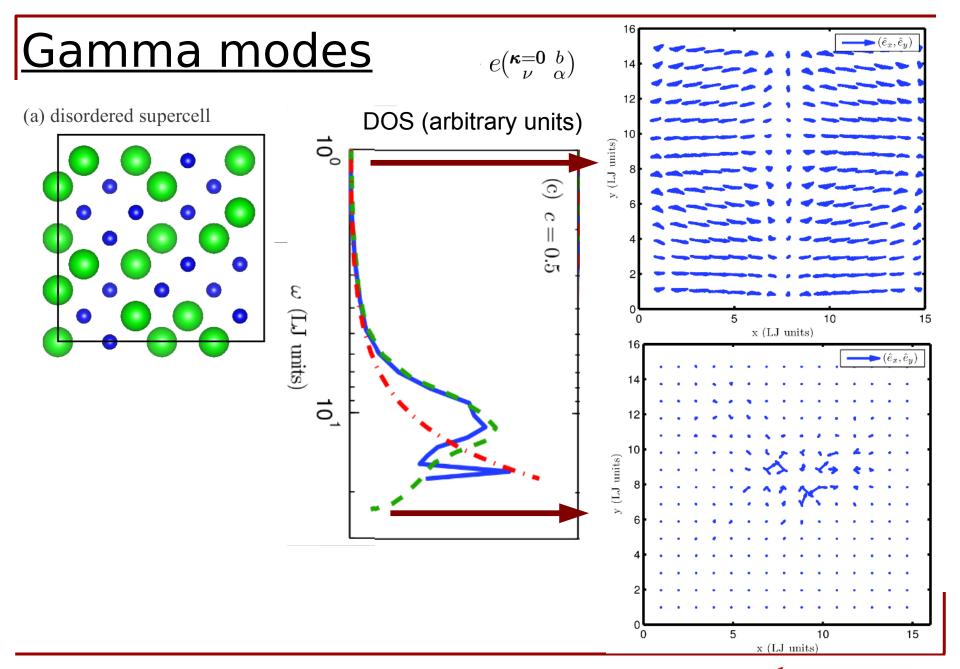
Thermal conductivity: SW silicon alloy

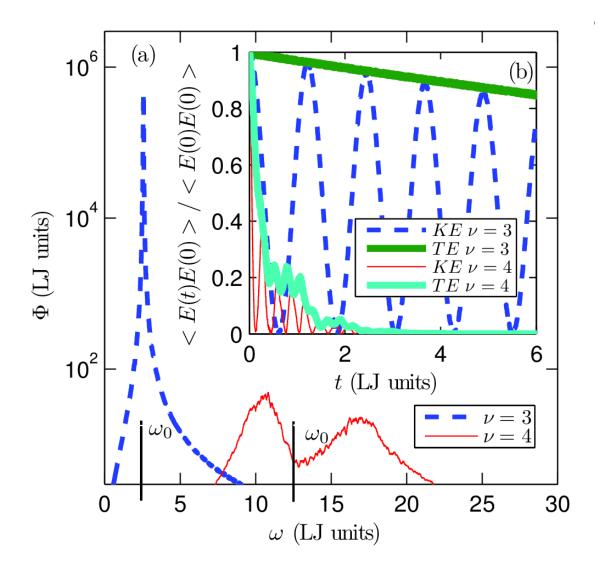




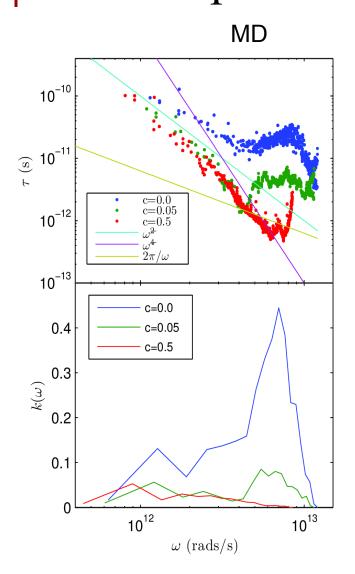


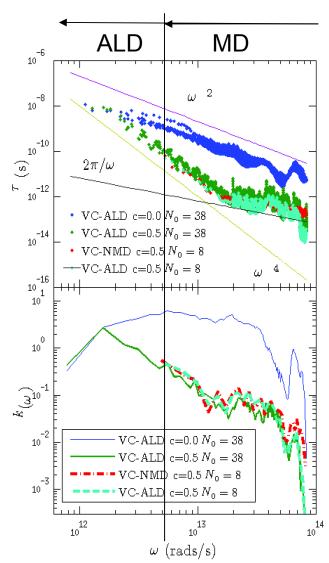






Phonon Spectrum: LJ Ar vs SW Si





MD-based:

1E4 modes

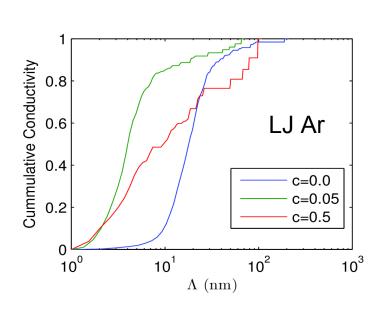
(7 days)*(100 cpu)

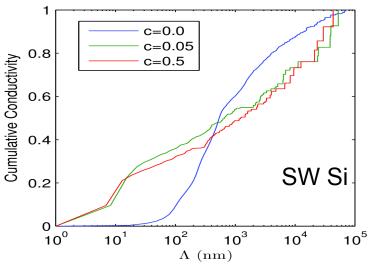
ALD:

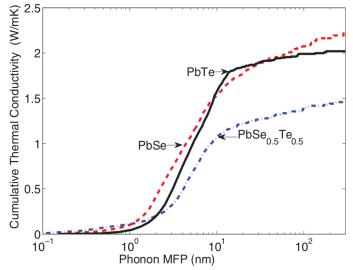
1E6 modes

(7 days)*(12 cpu)

Conductivity Accumulation







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