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# DISCUS

## Users Guide

Version 5.7

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written by

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# Preface

## Disclaimer

The DISCUS software described in this guide is provided without warranty of any kind. No liability is taken for any loss or damages, direct or indirect, that may result through the use of DISCUS. No warranty is made with respect to this manual, or the program and functions therein. There are no warranties that the programs are free of error, or that they are consistent with any standard, or that they will meet the requirement for a particular application. The programs and the manual have been thoroughly checked. Nevertheless, it can not be guaranteed that the manual is correct and up-to-date in every detail. This manual and the DISCUS program may be changed without notice.

DISCUS is intended as a public domain program. It may be used free of charge. Any commercial use is, however, not allowed without the explicit written permission of the authors.

## Acknowledgments

The parameters for the calculation of bond valence sums were compiled by I.D. Brown, McMaster University, Hamilton, ON, Canada.

## Using DISCUS

Publication of results totally or partially obtained using the program DISCUS should state that DISCUS was used and contain the following reference:

PROFFEN, TH. & NEDER, R.B. (1997) "DISCUS, a Program for Diffuse Scattering and Defect Structure Simulations" *J. Appl. Cryst.*, **30**, 171-175

## More information

This users guide can only provide program specific details. A broader discussion of simulation techniques and many DISCUS examples and macro files can be found in our book

NEDER, R.B. & PROFFEN, TH. "Diffuse Scattering and Defect Structure Simulations - A cook book using the programs DISCUS", *IUCr Texts on Crystallography*, Oxford University Press, 2008.

The macro files used in the cookbook can also be downloaded to the DISCUS homepage at <http://discus.sourceforge.net>.

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# Chapter 1

## Introduction

### 1.1 What is DISCUS ?

DISCUS (Proffen and Neder, 1997) is intended as a versatile tool to simulate crystal structures and the corresponding intensity distribution in reciprocal space. The program offers several features that enable the user to easily generate a structure and to introduce various defects. The main intend is to simulate defect structures, the program, however, is not limited in that respect. Ideal structures can be simulated as well and usually will form the basis from which the defect structure is developed.

The program can read a whole crystal or the asymmetric unit of a unit cell. The latter is expanded to the whole unit cell by use of the space group symbol. The definition of rigid molecules is supported by DISCUS. The structure can be stored as structure file or certain layers or projections of the crystal can be saved for graphical display. The program DISCUS is part of the diffuse program package. Several tools are available to modify single atoms or molecules within the structure or to alter the complete crystal. These tools include thermal displacements, waves, domains, stacking faults and generalised symmetry operations. One feature of DISCUS is the possibility to introduce correlated defects and distortions using Monte Carlo (MC) simulations. Various tools allow the analysis of a given defect structure including the calculation of the PDF of a given structure.

The Fourier transform segment of the program allows the user to calculate the intensity distribution along a line, arbitrarily oriented plane or a three dimensional volume through reciprocal space. The resolution in reciprocal space can be chosen by the user. The resulting intensity maps can be written in several output formats, to be displayed by standard visualisation programs, the program *KUPLOT* or be printed directly. The list of output formats includes the NeXus format. Neutron- as well as X-ray and electron scattering can be calculated. DISCUS supports the subtraction of the average structure factor and the calculation of the diffuse intensities as average over many small crystal volumes to create smooth noise free diffraction pattern. A detailed discussion is given in chapter 3. The direct analysis of measured diffuse scattering is possible using the Reverse Monte Carlo segment of the program. The program allows the refinement of the scattering intensity directly as well as the refinement of the PDF. The program uses a command language to interact with the user. No predefined requirement exists for a given sequence of commands. The commands can be typed at the DISCUS prompt, or read from a macro file. The command language includes a FORTRAN style interpreter that



allows the user to program loops and logical structures. Several real and integer variables as well as structural variables can be used to design the intended defect structure. On line help is provided which gives the user information on any of the DISCUS commands as well as examples for typical DISCUS sessions. More detailed information how to get further help is given in section 1.3.

## 1.2 News

This is a major new release of DISCUS.

The program now uses a flexible internal data storage. This allows flexible crystal sizes.

In the original version, centering operations were applied first, before the resto of the symmetry operations. Starting with DISCUS Version the normal symmetry operations are applied first and then the centering operations are applied.

A new connectivity concept allows faster Monte-Carlo refinements for non periodic objects.

The Monte-Carlo menu includes a purely repulsive energy term.

Structures can be stored internally to save disk I/O.

Kinematic Electron diffraction form factors have been added.

Import types now include CIF files and RMCprofile output files.

The new DISCUS SUITE combines DISCUS, KUPLOT, and DIFFEV into a single program, with many improvements for large computing facilities and parallel processing.

## 1.3 Getting started

After the program DISCUS is installed properly and the environment variables are set, the program can be started by typing 'discus' at the operating systems prompt. Information about the installation of DISCUS is given in the DISCUS package manual.

Symbol	Description
"text"	Text given in double quotes is to be understood as typed.
<text>	Text given in angled brackets is to be replaced by an appropriate value, if the corresponding line is used in DISCUS . It could, for example be the actual name of a file, or a numerical value.
text	Text in single quotes exclusively refers to DISCUS commands.
[text]	Text in square brackets describes an optional parameter or command. If omitted, a default value is used, else the complete text given in the square brackets is to be typed.
{text   text}	Text given in curly brackets is a list of alternative parameters. A vertical line separates two alternative, mutually exclusive parameters.

**Table 1.1:** Used symbols

The program uses a command language to interact with the user. The command `exit` terminates the program and returns control to the shell. All commands of DISCUS consist of a

Variable	Description
n[1]	Number of atoms within the crystal
n[2]	Number of different scattering types, i.e. atoms
n[3]	Number of atoms within the unit cell
n[4]	Number of molecules within the crystal
n[5]	Number of different molecule types
n[6]	Number of molecules within the unit cell
n[7]	Number of none-void atoms in the crystal
cdim[i,1]	Lowest coordinate of any atom (i=1,2,3 for x,y,z)
cdim[i,2]	Highest coordinate of any atom (i=1,2,3 for x,y,z)
env[i]	Index of neighbouring atoms after 'find' command
menv[i]	Index of neighbouring molecules after 'find' command
lat[i]	Lattice parameters (i=1..6 for a, b, c, $\alpha$ , $\beta$ , $\gamma$ )
vol[1]	Unit cell volume
rvol[1]	Reciprocal unit cell volume

Table 1.2: Crystal related variables

Variable	Description
m[i]	Number of scattering type (i.e. atom type) for atom i
b[i]	isotropic thermal factor B for atom <b>type</b> i
at_name[i]	Atom name for atom i
in_mole[i]	Molecule in which atom i is
prop [i]	Property of atom i
at_type[i]	Atom name for atom type i
x[i]	fractional x position of atom i
y[i]	fractional y position of atom i
z[i]	fractional z position of atom i
mol_cont[i,j]	Index of atom j in molecule i
mol_len[i]	Number of atoms in molecule i
mol_biso[i]	Isotropic B-value for molecule i
mol_dens[i]	Density of object i
mol_type[i]	Type of object / molecule / domain i
pdf_dens[i]	Number density for pdf calculations
pdf_scal[i]	Scale factor for pdf calculations

Table 1.3: Variables related to individual atoms and molecules

command verb, optionally followed by one or more parameters. All parameters must be separated from one another by a comma ",". There is no predefined need for any specific sequence of commands. DISCUS is case sensitive, all commands and alphabetic parameters MUST be typed in lower case letters. If DISCUS has been compiled using the -DREADLINE option (see installation files) basic line editing and recall of commands is possible. For more information refer to the reference manual or check the online help using ('help command input'). Names of

Type	Name	Description
real	bang(u1,u2,u3, v1,v2,v3 [,w1,w2,w3])	Returns the bond angle in degrees between <b>u</b> and <b>v</b> at site <b>w</b> . If <b>w</b> is omitted, the angle between direct space vectors <b>u</b> and <b>v</b> is returned.
real	blen(u1,u2,u3 [,v1,v2,v3])	Returns the length of the real space vector <b>v-u</b> . The vector <b>v</b> defaults to zero.
real	dstar(h1,h2,h3 [,k1,k2,k3])	Returns the length of reciprocal vector <b>k - h</b> in Å <sup>-1</sup> . Vector <b>k</b> defaults to zero.
real	rang(h1,h2,h3, k1,k2,k3 [,l1,l2,l3])	Returns the angle between reciprocal vectors <b>k - h</b> and <b>k - l</b> at site <b>k</b> . If <b>l</b> is omitted, the angle between reciprocal vectors <b>h</b> and <b>k</b> is returned.
real	scalpro(u1,u2,u3,v1,v2,v3 [, "dd"   "rr"   "dr"   "rd"])	Returns the scalar product between the two vectors <b>u</b> and <b>v</b> . Both vectors may be given in direct or real space coordinates, flagged in parameter no 7. "d" means real space, "r" reciprocal space.
real	vprod <u1>,<v1>,<w1>, <u2>,<v2>,<w2> [ ,<flag> ]	This is not a function but rather a command that calculates the vector product <b>1 X 2</b> . The <flag> is a string of three characters that indicates whether the input vectors or the output vector are given in direct or reciprocal space coordinates. The three characters correspond to: first input vector, second input vector resulting vector "drd" means: vector one is given in direct space coordinates, vector 2 is in reciprocal space coordinates (hkl) and the resulting vector product is to be given in direct space coordinates Any combination of "d" and "r" is allowed.

Table 1.4: Crystallographic functions

input or output files are to be typed as they will be expected by the shell. If necessary include a path to the file. All commands may be abbreviated to the shortest unique possibility. At least a single space is needed between the command verb and the first parameter. No comma is to precede the first parameter. A line can be marked as comment by inserting a "#" as first character in the line.

The symbols used throughout this manual to describe commands, command parameters, or explicit text used by the program DISCUS are listed in Table 1.1. There are several sources of information, first DISCUS has a build in online help, which can be accessed by entering the command `help` or if help for a particular command <cmd> is wanted by `help <cmd>`. This manual describes background and principle functions of DISCUS and should give some insight in the ways to use this program.

As we have mentioned before, DISCUS is controlled by a powerful command language which is common to all programs in the DISCUS package. A detailed description of the command language can be found in the DISCUS Reference Guide which is included in the distribution.

Variables specific to the program DISCUS are shown in Tables 1.2 and 1.3. DISCUS also provides a number of structure related functions which are listed in Tab. 1.4.

## Chapter 2

# Creating structures

The first steps towards any simulation are to create the desired structure and possibly to modify it. DISCUS offers two different ways to perform this task, the structure can either be generated from the contents of a asymmetric unit of a unit cell or be completely read from a file. For each atom the program stores its type, its fractional coordinates ( $x,y,z$ ) an isotropic thermal coefficient ( $B$ ) and a set of property flags. If the crystal is generated from an asymmetric unit, the program uses the space group symbol read from the data file to generate the unit cell from the asymmetric unit. Prior to this generation, the space group symbol is checked for consistency with the lattice constants. An error message flags any inconsistencies. The unit cell is copied to generate a crystal of desired (rectangular) dimensions.

The following sections describe the structure file formats and how to group atoms to (rigid) molecules. Once a crystal (which could be as small as one atom) is generated, several tools are provided to modify the crystal. These tools can be grouped into those that affect individual atoms and those that affect the whole crystal. The first group includes the following modifications of an individual atom: moving, replacing, removing and inserting new atoms. These are discussed in chapter 5. The second group includes at present the following modifications of the crystal at large, for more details see chapter 6: thermal displacement, waves, stacking faults and domains, generalized symmetry operations ( chapter 4), Monte-Carlo modelling ( chapter 8), and Reverse-Monte-Carlo modelling ( chapter 10).

### 2.1 Reading structure files

A unit cell or a whole structure is read from file by the command `read`. The format of these two file types is identical. If a unit cell is read, the contents of the file is regarded as the asymmetric unit of the unit cell. The space group information is used to generate the whole unit cell. If, on the other hand, a structure is read, the contents of the file is taken as it is.

A keyword controlled structure file format is used allowing more flexibility, e.g. the use of molecules. However, DISCUS is capable to read both the old and the new structure files automatically recognising the correct format. Furthermore the command `format nokey` in the `save` segment allows the user to save a structure using the old format (see section 2.4). As the non keyword file format lacks many of the storage features its use is highly discouraged.

### 2.1.1 Structure file format

The standard DISCUS structure file is a simple text file starting with a section with keywords setting parameters like lattice constants or space group followed by a section with the actual atoms. The sequence of keywords is arbitrary with two exceptions, the first line **must** contain the keyword `title` setting a title and the last keyword **must** be `atoms` followed by a list of atoms and/or molecule definitions (see section 2.1.2). Each atom within the asymmetric unit or complete structure is defined by its name (e.g. ZR), the fractional coordinates  $(x, y, z)$  and an isotropic temperature factor  $B$ . A list of valid keywords within a structure file is given in Table 2.1.

Keyword	Description
#	Allows comments in the keyword section of the structure file..
adp	A list of isotropic B values for all atom types. If present, these values will override the parameter specified for an atom upon reading a structure.
atoms	Marks the start of the atom list and must be the <b>last</b> keyword except for 'molecule'.
cell	Parameters $a, b, c, \alpha, \beta, \gamma$ define the lattice constants of the crystal. Note that <b>all</b> six parameters must always be given.
generator	Allows to define generators additional to the ones defined by the space group. The 12 parameters define the symmetry operation.
molecule	Defines rigid molecules. The keyword must stand in the atoms section of the structure file (see section 2.1.2).
ncells	Parameters $nx, ny, nz$ and $nc$ define the size of the crystal in unit cells and the number of atoms within a unit cell. This information is needed if a complete structure is read in order to allow DISCUS to determine the crystals size.
scat	A list of atom types. This will ensure that the atom types have a fixed sequence.
spcgr	Sets the space group for the crystal.
symmetry	Defines an additional symmetry operation defined by the 12 parameters given.
title	Sets a title for the structure file. This <b>must</b> be the first keyword in the file.

Table 2.1: List of keywords for structure files

Keywords are processed the same way as normal commands and parameters have to be separated by commas. An example input file for cubic zirconia ( $ZrO_2$ ) is shown below.

```

title Structure of cubic ZrO2
spcgr Fm-3m
cell 5.14, 5.14, 5.14, 90., 90., 90.
atoms
ZR 0.00000, 0.00000, 0.00000, 0.5
O 0.25000, 0.25000, 1./4., 1.0

```

The first line is the required `title` line describing the structure. The next line specifies the space group  $Fm\bar{3}m$ . The symbols used should be the Hermann-Mauguin symbols used in

International Tables for Crystallography Vol. A Wilson et al. (1983). A center of inversion as in this example should be given as - sign immediately preceding the axis. Lattice types need to be given as capital characters, mirror planes as small characters. Monoclinic cell choices 2,3 or unique c-axis will be assumed if the corresponding non standard Hermann-Mauguin symbol is used. DISCUS checks the given space group symbol for contradictions with the lattice constants and in case of an error the unit cell is not read. A complete list of valid space group symbols is part of the online help and can be accessed via 'help space' from the DISCUS command line. The next line in the example above gives the lattice constants of  $a = b = c = 5.14\text{\AA}$  and  $\alpha = \beta = \gamma = 90$  degrees. Note that DISCUS requires all six values to be given. The keyword 'atoms' in the example file which must be the last keyword starts the section with the list of atoms. Here zirconium occupies site 4(a) on (0,0,0) and oxygen is on 8(f) at  $(\frac{1}{4}, \frac{1}{4}, \frac{1}{4})$ . The isotropic temperature factors for Zr and O are set to  $0.5\text{\AA}^2$  and  $1.0\text{\AA}^2$  respectively.

If the lattice parameters or the atom coordinates are separated by comma, you can use simple expressions like 1./2. or even variables instead of explicit numerical values.

Additional generators can be defined through the optional `generator` keyword. These generators act identical to the generators defined through the space group symbol. All previously generated copies of the atoms in the asymmetric unit are copied by this generator, and will in turn be copied by any generators following later. Since these additional generators are applied after the space group generators, you can use these generators to create non-standard groups or to create a set of symmetries that does not form a group. The syntax of the 'generator' keyword is as follows:

```
generator g11,g12,g13,g14, g21,g22,g23,g24, g31,g32,g33,g34
```

Copies of an atom at (x,y,z) will be calculated using the following equation:

$$\begin{pmatrix} x' \\ y' \\ z' \end{pmatrix} = \begin{pmatrix} g_{11} & g_{12} & g_{13} \\ g_{21} & g_{22} & g_{23} \\ g_{31} & g_{32} & g_{33} \end{pmatrix} \cdot \begin{pmatrix} x \\ y \\ z \end{pmatrix} + \begin{pmatrix} g_{14} \\ g_{24} \\ g_{34} \end{pmatrix} \quad (2.1)$$

Additional symmetry operations can be defined through the optional `symmetry` keyword. These symmetry operations act different than the generators described above which are defined through the space group symbol or listed as additional generators. The symmetry operations copy only those atoms created by the generators. In contrast to generators they do not act on copies of the atoms created by previous symmetry operations. Both keywords `generator` and `symmetry` define the symmetry operation in a similar way using 12 parameters as shown in equation 2.1

The following example shall illustrate the difference between generators and additional symmetry operations. The following two generators

```
generator 1,0,0,0.5, 0,1,0,0.5, 0,0,1,0.0
generator 1,0,0,0.5, 0,1,0,0.0, 0,0,1,0.5
```

would create the following copies of an atom at (0,0,0):  $(\frac{1}{2}, \frac{1}{2}, 0)$ ,  $(\frac{1}{2}, 0, \frac{1}{2})$  and  $(0, \frac{1}{2}, \frac{1}{2})$ . In contrast similar symmetry operations

```
symmetry 1,0,0,0.5, 0,1,0,0.5, 0,0,1,0.0
symmetry 1,0,0,0.5, 0,1,0,0.0, 0,0,1,0.5
```



will only generate the following two copies of an atom at (0,0,0):  $(\frac{1}{2}, \frac{1}{2}, 0)$  and  $(\frac{1}{2}, 0, \frac{1}{2})$  since the symmetry operations will not act on previously generated copies of the atom at (0,0,0). The second symmetry operation copies only the atom at (0,0,0), not the atom at  $(\frac{1}{2}, 0, \frac{1}{2})$ , since this atom was created by the previous symmetry operation.

### 2.1.2 Using molecules

The keyword controlled structure file format of DISCUS allows the definition of molecules using the `molecule` keyword. This keyword is allowed anywhere between the atoms of the unit cell file. It marks the beginning of a group of atoms that are grouped to form a molecule. The individual atoms are listed in the usual way (see section 2.1.1). The keyword `molecule end` signals the end of a molecule. All atoms still listed in the unit cell file are treated as individual atoms. The molecule related keywords are listed in table 2.2.

Keyword	Description
<code>molecule</code>	Defines the start of a molecule.
<code>molecule atoms</code>	Lists atom numbers belonging to current molecule.
<code>molecule biso</code>	Defines an overall B-value for the current molecule.
<code>molecule character</code>	Defines the character of a molecule.
<code>molecule content</code>	Defines the start of particular molecule type.
<code>molecule density</code>	Defines the density of molecules with "object" character.
<code>molecule end</code>	Defines end of a molecule atom list.
<code>molecule generator</code>	Starting with DISCUS 3.6 the following keywords have become obsolete. Defines generators for the internal symmetry of the molecule.
<code>molecule symmetry</code>	Defines internal symmetry operations for the molecule.

Table 2.2: List of keywords molecules

Starting with DISCUS 3.6, the internal symmetry of a molecule is automatically determined from the Wyckoff position of the first atom. It is mandatory that this first atom is at a Wyckoff position, whose local site symmetry corresponds to the molecule symmetry. If the molecule does not possess an atom at this site, insert a "VOID" at the site.

The internal symmetry of the molecule can be specified using the 'generator' and 'symmetry' sub-keywords. The generators are internal symmetry operations of the molecule. DISCUS compares the lists of atoms created by the space group and by the molecule generators. Identical sections are linked to one molecule. Atoms created by other symmetry operations, e.g. lattice centering will form a new molecule of the same type. The generators of the molecule symmetry should be the generators that create the site symmetry. See the section on site symmetry in the International Tables A Wilson et al. (1983) for further details. As in the previous section, symmetry operations will only act on the *original* atoms of the molecule whereas generators will operate on previously generated copies of atoms as well. There is one **important restriction** how molecules are defined in DISCUS: The first atom of any molecule defines the origin of the molecule used by various subsequent commands. In case the origin lies on a symmetry element of the space group it must be located at the point of highest symmetry of the molecule.



If the structure does not have an atom at this site you must include a *void* on this site. This could be the case e.g. if you have an empty triangle on a threefold axis.

Alternatively to defining a molecule as discussed above, the command `molecule content` and `molecule atoms` might be used to define molecule types and the corresponding list of atom indices belonging to that molecule. This procedure is used to be able to save structures containing molecules since the order of the atoms required by various DISCUS functions might prevent storing atoms in groups belonging to a particular molecule. Check section 2.4 for more details about saving structures.

## 2.2 Freely generated structure

As an alternative, DISCUS can create a structure freely by the `free` command. This alternative is helpful to avoid disk I/O.

```
1 read
2 free 5.14, 5.14, 5.14, 90.00, 90.00, 90.00, Fm-3m
3 insert Zr, 0.0000, 0.0000, 0.0000, 0.1
4 insert O , 0.2500, 0.2500, 0.2500, 0.1
```

This example illustrates the alternative style to create the asymmetric unit of cubic zirconia. The last parameter in line 2 instructs DISCUS that the freely read structure should belong to space group  $Fm\bar{3}m$ . This parameter is optional and defaults to P1. After these four commands DISCUS has created just the atoms within the asymmetric unit. To build a full crystal, save this asymmetric unit (optionally save it as an internal file) and read it again with the `cell` command. See section 2.4 for more information on the save procedure.

## 2.3 Internal storage of a structure

All atoms are stored sequentially in an array. If a complete structure is read using the `stru` command, the atoms are stored in the same sequence as found on the input file. If several unit cells have been generated by the command `cell` in the `read` sublevel, the contents of all unit cells are stored consecutively. The `cell` command always generates a rectangular block of  $nx, ny, nz$  unit cells along the direct axes. The innermost, fastest loop is over  $x$ , the outermost, slowest loop is over  $z$ .

Within each unit cell, the atoms are stored in the sequence that they have been read from file. Each atom is multiplied by the symmetry generators of the respective space group. The sequence of generators has been chosen identical to that in the International Tables for Crystallography Wilson et al. (1983), Vol. A. The sequence of symmetrically equivalent atoms is therefore identical to that of the International Tables.

The generators of a space group are applied in the same order as given in Vol A of the International Tables. Thus the generated atoms will be in the same sequence as printed for each Wyckoff position of each space group.

If the space group is centered, the generators that correspond to symmetry operations are applied first and then each block is repeated by application of the centering generators. This is illustrated by the following example:

*Space group C2 No.5, Wycoff position 4(c)  $x,y,z$*

*The atoms on this site are*

$(0, 0, 0; \frac{1}{2}, \frac{1}{2}, 0) +$

$x, y, z; \bar{x}, y, \bar{z}$

DISCUS stores these four atoms on position 4(c) in the following sequence:

$x, y, z$

$\bar{x}, y, \bar{z}$

$x + \frac{1}{2}, y + \frac{1}{2}, z$

$\bar{x} + \frac{1}{2}, y + \frac{1}{2}, \bar{z}$

Prior to DISCUS 3.6, in accordance with the header lines in the International Tables, the generators required for centered space groups were applied first. The atoms generated by these generators immediately followed each position generated by the ordinary symmetry operators as illustrated for the following example:

$x, y, z$

$x + \frac{1}{2}, y + \frac{1}{2}, z$

$\bar{x}, y, \bar{z}$

$\bar{x} + \frac{1}{2}, y + \frac{1}{2}, \bar{z}$

The old sequence caused problems for molecules in rhombohedrally centered space groups. In order to provide backward compatibility, the `define` command allows you to switch between the newer storage sequence and the older one.

The origin within the model crystal is taken as the center of the simulated structure, i.e. the coordinates of an atom on site (0,0,0) of a crystal 20x20x20 unit cells in size would range in  $x, y$  and  $z$  from -10.0 to 9.0.

If the chemistry or position of an atom is modified, its position within the structure array remains the same. If it is deleted, it is replaced by atom type VOID. The position of all other atoms remains unchanged. This feature enables the user to calculate the position of an atom in the structure array, irrespective of any changes that took place in the structure. All new atoms are added at the end of the structure array. Their sequence is entirely based on the sequence of generation.

The command `trans` in the `chem` sublevel of DISCUS allows to transform between the index of an atom and its unit cell and site number. Alternatively the atom index  $l'$  for an atom on site  $l$  in unit cell  $i, j, k$  can be calculated using the following equation:

$$l' = \{(k-1) \cdot nx \cdot ny + (j-1) \cdot nx + (i-1)\} \cdot n[3] + l \quad (2.2)$$

Here  $nx, ny, nz$  are the crystals dimensions in unit cells and  $n[3]$  (see table 1.2 in section 1.3) contains the number of atoms per unit cell. Some calculations of DISCUS (e.g. usage of lots in Fourier sublevel, quick mode for bond length calculations, ..) require this particular order of storage of the atoms. However, in order to save CPU time for the Fourier transform and disk space when saving a structure, the `purge` command may be applied which will delete all empty sites in the crystal and the given relation between atom index and unit cell/site is invalid. The authors generally do **not** recommend the usage of the 'purge' command.

## 2.4 Saving structures

A particular structure stored in DISCUS can be saved to a file using the `save` command. In cases where `save` is followed by a filename, the structure is directly saved using the currently set options.

If `save` is entered alone, DISCUS will enter a `save` sublevel which allows the user to select various options and select all or only particular atoms to be saved. Note that the settings are valid for all subsequent calls of `save filename`. Generally the user can choose between the old structure file format (see section 2.1) and the new keyword controlled structure file format, which is the recommended format to be used. Beside the required keywords the user is able to select which additional keyword shall be written to the file. Note that in cases where a structure containing molecules is saved without selecting the `molecule` keyword, the molecule information of the crystal will be lost. Check the online help for `save` for a complete list of options.

Starting with DISCUS Version 5.0 you can save a structure into an internal memory location within the CPU instead of onto the disk. Simply prepend the filename with the characters "internal". DISCUS will allocate memory to store the structure. To recover the structure read it with `read cell` or `stru`. The filename must of course start again with "internal". This internal storage is highly recommended for temporary or intermediate structures that you do not really need. This holds especially during a refinement on a high performance computer cluster.

## 2.5 Exporting structures

The structure, or parts thereof can be exported to file in formats suitable for display by corresponding programs. Currently supported programs are KUPLOT which is part of the DISCUS package and a number of structure plotting packages such as *ATOMS* and *DIAMOND*. A simplified *CIF* file can be read by most crystallographic packages. Check the online help for an up-to-date list of supported programs.

Different export options are available in the `plot` sublevel of DISCUS. Individual atom or molecule types to be included in the output might be selected. The extent of the crystal to be exported as well as an optional limitation to a given slice within the crystal allow the user to select specific regions within the crystal. Such a slice is defined by a point  $v$  in real space its thickness and its normal. For each atom in the crystal the vector from the point  $v$  to the atom is projected onto the real space normal to the slice. If the length of this projection is less than the thickness, the atom is plotted, else not. DISCUS simply writes the coordinates of selected atoms. The sequence of indices in the output file can be defined by the user as any permutation of  $x,y,z$ ;  $x,z,y$ ;... The output for KUPLOT will result in a projection of the structure along one of the crystallographic axes or a projection along the normal of the slice onto the plane, while the *GNUPLOT* format can be used for a three dimensional viewing. The file format for KUPLOT includes marker type, colour and size for the individual atom or molecule representation. These properties can be altered using the `set` command within the `plot` sublevel. Assuming the structure has been written to file *atoms.xy*, the following KUPLOT macro would be appropriate:

```
load cr,atoms.xy    ! loading the data file
plot                ! plot it
```

For more information about the program KUPLOT refer to the KUPLOT manual or online help. To import into a CIF file viewer export with the program `cif` option:

```
plot
  program cif
  sel      all
  outfile your_name.cif
  run
exit
```

## 2.6 Importing structures

DISCUS can import crystal structures from a few external file formats, as listed in Table 2.3. The relevant commands are found in the `import` menu. DISCUS will read the import structure and automatically save a DISCUS type file, whose extension has been replaced by **stru**.

Format	Description
CIF	The Crystallographic Information File format, as described in the International Tables Wilson et al. (1983) Vol. F.
RMCPROFILE	Crystal structure storage format as used by RMCProfile
SHELX	Import a res or ins file as used by SHELX programs
CRYSTALMaker	Crystal structure storage format as used by CrystalMaker

Table 2.3: List of supported import file formats

## 2.7 Properties

Within a crystal you will typically have a lot of different atom types. In addition to the type (Fe, Mn, Cl ...) atoms may be distinguished by further properties. An atom might be within a molecule or be a separate atom. DISCUS labels these different properties as logical flags and updates these automatically if any changes occur. Many of the modification and output tool can utilize these properties to include or reject a specific atom. The properties are:

## 2.8 Connectivity

Many of the tools in DISCUS need to reference neighboring atoms around a given atom. It is very time consuming to search throughout the entire crystal if an atom fulfills the desired neighborhood conditions. To avoid these length loops DISCUS provides two styles to reference a neighboring atom.

The first style is the fastest but it depends on the initial storage order of the atoms, which was described in section 2.3. As long as this sequence is maintained DISCUS can reference a neighbor by a vector relationship:

```
1  set vec, 1, 1, 1, 1, 0, 0
2  set vec, 2, 3, 4, 1,-2, 4
```

Keyword	Description
Normal	The atom is a normal atom or a void.
Molecule	The atom is part of a molecule
Domain	The atom is within a domain.
Outside	The atom is outside the crystal, i.e. on the other side of a limiting surface. Usually these atoms are voids as well.
External	The atom is within a user defined distance to an external boundary, either on the inside or outside.
Internal	The atom is within a user defined distance to an internal boundary, either on the inside or outside. Internal boundaries arise through domains.
Ligand	The Atom is part of a ligand molecule that has been placed on top of the crystal.

Table 2.4: List of property flags

The first parameter after the `vec` keyword numbers the vector relationships. The next two parameters describe which atom suites with a set of unit cells is related. The last three parameters describe the relative shift of the unit cell in which the second site, the neighbor, is found. Line (1) would thus create a relationship between any atom on site one in a unit cell to the atom on site 1 in a unit cell shifted by  $[1,0,0]$ . Line (2) creates a relationship between an atom on site 3 in a unit cell and an atom on site 4 in a unit cell shifted by  $[1,\bar{2},4]$  unit cells. These relationships hold for all pairs throughout the crystal. In order to quickly calculate which atom in our internal sequence is referenced, the atoms must still be in their original order.

If you add a lot of atoms, or if you shape the crystal with the boundary commands (see section 6.5) and remove unnecessary outside voids, the internal sequence of atoms may change dramatically. Under these circumstances the vector relationships may not work efficiently any longer. For that reason you will find several warnings with respect to the `purge` command. To allow a fast reference to neighboring atoms DISCUS offers the `connectivity` menu. This menu allows you to define relationships between any pair of atoms through distance intervals:

```

1  connectivity
2  reset
3  add Zn, O , 1.90, 2.10, zn_o_short
4  add Zn, Zn, 3.10, 3.30, zn_zn_second
5  add O , Zn, 1.90, 2.10, o_o_short
6  add O , O , 3.10, 3.30, o_o_second
7  create
8  show
9  exit
10 ...
11 connectivity
12 recreate Zn, zn_zn_second
13 exit
```

In this example we clear all previous connectivity lists in line (2) and add two connectivities. The first (line 3) relates Zn and O atoms that are within the interval 1.90 to 2.10 Å and provides a name `zn_o_short` for this relationship. This relationship is uni directional, Zn is always at the center of the connectivity polyhedron. Thus to group all pairs, lines 5 and 6 add the opposite relationships with the oxygen as central atom.

The `add` command simply creates the definitions, the actual lists are generated with the `create` command. As the program has to perform a double loop over all atom pairs this may take a bit of time, but you will easily regain this time during a Monte-Carlo simulation.

In line 12 we recreate the connectivity `zn_zn_second` around Zinc atoms. This might be necessary, if atoms have been moved around.

## Chapter 3

# Fourier transform

### 3.1 Calculating scattering intensities

Basically DISCUS calculates the Fourier transform (neutron, X-ray, or electron) according to the standard formula for kinematic scattering given in equation 3.1.

$$F(\mathbf{h}) = \sum_{i=1}^N f_i(\mathbf{h}) e^{2\pi i \mathbf{h} \mathbf{r}_i} \cdot e^{-\frac{B|\mathbf{h}|^2}{4}} \quad (3.1)$$

The sum is over all  $N$  atoms in the crystal, where  $f_i$  is the atomic form factor (or scattering length in case of neutrons),  $\mathbf{r}_i$  the fractional coordinate of the atom. The sum is calculated at all points  $\mathbf{h}$  in reciprocal space. The form factors are tabulated and calculated once at each  $\mathbf{h}$  for all species present in the crystal. Optionally the Debye-Waller factor is calculated. Isotropic  $B$  are used for the representation of thermal disorder. The realisation of the actual code to calculate the Fourier transform is based on the program *DIFFUSE* (Butler and Welberry, 1992). By limiting the calculation to an equidistant grid and splitting the sum into sums over equal atom types, the computing time required dropped by a factor of 4 to 6 (depending on compiler and hardware) compared to calculating the sum given in equation 3.1 in a straight forward way. More details about the algorithm used can be found Butler and Welberry (1992). There are two considerations when calculating the scattering intensities: Unwanted scattering contributions due to the finite size of the model crystal and the fact that the complete model scatters coherently, which leads to unwanted high frequency oscillations in the calculated pattern. Finite size effects are avoided by using a grid of  $n/N$  where  $N$  is the number of unit cells. Coherence might be limited by using the `lots` command in the `four` module (see online help). A detailed discussion of these topics can be found in Neder and Proffen (2007).

Let us consider a simple example. Assuming a structure has been read or created, the scattering pattern can be calculated using these commands:

```
1 four
2 xray
3 wvle moal
4 ll 0.0,0.0,0.0
5 lr 2.0,0.0,0.0
6 ul 0.0,2.0,0.0
7 na 101
8 no 101
```

```
9   set aver,0.0
10  lots off
11  run
12  exit
```

After entering the Fourier segment of DISCUS (line 1),  $MoK\alpha$  radiation is selected in lines 2 and 3. Next the lower left, lower right and upper left corner of the desired plane in reciprocal space are specified (lines 4-6). The number of grid points in both directions are set in lines 7 and 8. Finally the subtraction of the average scattering intensity (Bragg peaks)  $\langle F \rangle$  is disabled (line 9) and the lot option is switched off (line 10). Then the Fourier transform is calculated (line 11). Note that one needs to save the result in the output segment of DISCUS as we will discuss later in this chapter.

Starting with DISCUS 5.0 you can also calculate a 3-dimensional volume in reciprocal space. A third axis is added to the values defined in the previous macro:

```
1  four
2  xray
3  wvle moal
4  ll 0.0, 0.0, 0.0
5  lr 2.0, 0.0, 0.0
6  ul 0.0, 2.0, 0.0
7  tl 0.0, 0.0, 2.0
8  na 101
9  no 101
10 nt 101
11 set aver,0.0
12 lots off
13 run
14 exit
```

This third axis defines the top left corner of the reciprocal space volume. The corresponding data can be saved as a series of slices parallel to the layer defined by the corners ll, lr, ul or as a 3D file in the NeXus file format or as a series of 2D slices.

## 3.2 Powder diffraction

In addition to the calculation of single crystal scattering intensities, DISCUS can also compute a powder diffraction pattern of a given structure. There are two ways DISCUS can calculate powder intensities: One is a complete integration of reciprocal space which is then mapped onto the desired range between  $2\theta_{min}$  and  $2\theta_{max}$ . The second algorithm uses the Debye scattering equation to calculate the powder pattern. More details as well as examples are given in our book (Neder and Proffen, 2007). Information about the related commands are found in the online help using the command `help powder`.

## 3.3 Fourier methods

Additional to the Fourier transform of a real space structure, three other Fourier transforms are available in the current version of DISCUS : *difference Fourier*, *inverse Fourier* and calculation of



the *Patterson* (Fourier transform of scattering intensities). All three of these Fourier transforms use the following equation:

$$\rho(\mathbf{r}) = \frac{1}{V} \sum_{i=1}^N F_i(\mathbf{h}) e^{-2\pi i \mathbf{h} \cdot \mathbf{r}_i} \quad (3.2)$$

The resulting density function  $\rho(\mathbf{r})$  and the Fourier coefficient  $F(\mathbf{h})$  take different meanings depending on the intended inverse Fourier transformation listed in table 3.1. The Fourier co-

Command	$F(\mathbf{h})$	$\rho(\mathbf{r})$
diff	$F_{obs} - F_{calc}$	difference scattering density
inv	$F_{obs}$	scattering density
patt	$I_{obs}$	Patterson density

**Table 3.1:** List of available Fourier methods

efficient are in general complex numbers. Therefore, DISCUS requires in most cases two input files for the inverse Fourier transforms. The allowed combinations are listed in the help file and the command reference.

### 3.4 Output file formats

All results of the Fourier transforms are written at the output segment of DISCUS. The following values can be saved:

*intensity, amplitude, real part, imaginary part and phase angle (in degrees)*

The Fourier transform calculates the real and imaginary part, all other values are calculated at the time of output. If the average structure factor  $\langle F \rangle$  was subtracted during the calculation of the Fourier transform, the corresponding values of  $\langle F \rangle$  can be saved as well. Note that if lots were used to calculate the Fourier transform, only the intensity values can be saved to a file. There is no need to calculate the Fourier again, if several of the values listed are to be written to file. Just define a new output value and output file name and run the output again. The desired output format is selected using the `format` command. The available output formats are listed in the table 3.2. The standard and gnu output formats write the actual numbers calculated by the Fourier transform. The KUPLOT or standard file format is defined as:

```
1      na,no
2      xmin,xmax,ymin,ymax
3ff    z,z,z,z,z
```

The values 'na' and 'no' given in the first line define the size of the section and are identical to the values given for the 'na' and 'no' commands at the Fourier segment. KUPLOT uses the coordinates along the abscissa and ordinate to scale the resulting picture. Since only two pairs of coordinates are read, the user has to define the necessary indices. 'xmin' and 'ymin' are the 'x' and 'y' coordinates of the lower left corner in reciprocal space, 'xmax' the 'x' coordinate of the lower right corner and 'ymax' the 'y' coordinate of the upper left corner. Which of the indices, h,k or l is interpreted as 'x' and 'y' coordinate depends on the values given for the 'abs' and 'ord' commands at Fourier sublevel. If for example the  $(hkl) \cdot [1\bar{1}0] = 0.0$  layer is calculated,

Output format	Description
stan	Output is saved as real numbers in a format suitable for KUPLOT which is part of the DISCUS program package.
mrc	Create output in the MRC file format.
nexus	Output is saved in the NeXus file format. See <a href="http://nexus.org">nexus.org</a> for information on the NeXus file format and file viewers.
gnu	Real numbers suitable for the program <i>GNUPLOT</i> which can be obtained on most software archive sites.
pgm, ppm	Integer bitmaps in PNM format as defined by Jef Poskanzer (gray scale and colour). Various programs are capable of reading PNM files and the <i>pnmplus-package</i> is a freely available collection of tools and conversion programs from PNM to virtually any other graphics format.
post	Creates a Postscript bitmap suitable for direct printing or to be imported by other programs.

Table 3.2: Output formats for scattering intensities

suitable values would be 'abs h' and 'ord l'. Now the h index of the lower left corner is written as 'xmin', and the l index as 'ymin'. The values are written row by row, each row consisting of the values along the abscissa. An empty line separates the rows in the output file.

The *GNUPLOT* output is written row by row, each data point within each row in its own line of output. The format of one such row is: " $h_1, h_2, z - value, h_3$ ". Again the sequence of indices h,k and l depends on the values given for the commands *abs* and *ord* at Fourier sub menu. For the bitmap output the calculated value is scaled linearly to values between 0 and 255. All values less than a definable threshold using the 'thresh' command are set to zero, all values above a maximum threshold are set to 255. Again, the bitmap is written row by row. A color map included in file *color.map* is used to attribute colors to the output values.

## Chapter 4

# Crystallography

In this chapter crystallographic calculations such as conversion from direct to reciprocal space and *vice versa* are discussed as well as symmetry operations and unit cell transformations. The mathematical background of the described calculations can be found e.g. in Sands (1995).

### 4.1 Crystallographic calculations

In order to determine the proper place of a new atom, or to check existing atoms several functions perform calculations that are based on the metric of the crystal. Several intrinsic functions are provided by DISCUS. They are listed in Table 1.4 in section 1.3 of this manual. Those function allow the user to calculate e.g. bond length and bond angles within the current crystal.

Command	Description
d2r	Conversion from direct (real) space to reciprocal space
r2d	Conversion from reciprocal space to direct space
vprod	Vector product in real or reciprocal space
proj	Calculates the projection from one vector on another in real or reciprocal space

**Table 4.1:** Commands for crystallographic calculations

The crystallographic calculation commands provided by DISCUS are shown in Table 4.1. All commands display the results on the screen and store them in the variable `res[i]` (see section Variables in the package manual), which allows further use of the results in macro files.

The commands `d2r` and `r2d` calculate the components of a given vector  $\mathbf{u}$  in the complementary space. The vector product  $\mathbf{w} = \mathbf{u} * \mathbf{v}$  gives a vector  $\mathbf{w}$  that is normal to the two vectors  $\mathbf{u}$  and  $\mathbf{v}$ . The length of the resulting vector  $\mathbf{w}$  is given by  $|\mathbf{u}| \cdot |\mathbf{v}| \cdot \sin(\angle(\mathbf{u}, \mathbf{v}))$  which is the area of the parallelogram spanned by the vectors  $\mathbf{u}$  and  $\mathbf{v}$ . This vector product is calculated by the command `vprod` and any of the three vectors  $\mathbf{u}$ ,  $\mathbf{v}$  and  $\mathbf{w}$  may be in direct or reciprocal space. The command `proj` computes the projection of a vector  $\mathbf{u}$  onto the vector  $\mathbf{v}$  or on the plane normal to vector  $\mathbf{v}$ . The length of the projected vector is given by  $\mathbf{u} \cdot \mathbf{v} / |\mathbf{v}|$ . Again all input or resulting vectors can be in direct or reciprocal space. More details about these commands can be found in the online help of DISCUS. This version of DISCUS does not offer a special

command for the scalar product of two vectors. You can readily calculate the scalar product by keeping in mind its definition:  $|\mathbf{u}| \cdot |\mathbf{v}| \cdot \cos(\angle(\mathbf{u}, \mathbf{v}))$ . You can calculate this value by the expression: `blen(u)*blen(v)*cosd (bang(u, v))`.

## 4.2 Generalised symmetry operations

Lets assume that you want to rotate the oxygen atoms of a  $\text{SiO}_4$  tetrahedra around one of the Si-O bonds. Or you want to create the twinned structure of a triclinic crystal. The operations require the application of a symmetry element whose axis is not parallel to any of the base vectors and/or whose rotation angle is different from 60, 90, 120, or 180°. Needless to say, the resulting symmetry matrix will not contain just one's and zero's. A general symmetry operation can be described by a 4x4 matrix of the following form:

$$S = \begin{pmatrix} w_{11} & w_{12} & w_{13} & t_1 \\ w_{21} & w_{22} & w_{23} & t_2 \\ w_{31} & w_{32} & w_{33} & t_3 \\ 0 & 0 & 0 & 1 \end{pmatrix} \quad (4.1)$$

Here the components  $w_{ij}$  define the rotation and  $t_i$  the translational part of the symmetry operation. A detailed description for the derivation of this general symmetry operation is found in Sands (1995).

Additionally DISCUS allows one to select the atom or molecule types to be included in the symmetry operation using the `sele/dese` and `mse/mdes` commands. The user can specify whether the atom or molecule created by the symmetry operation replaces its original or is inserted as a new atom or molecule in the crystal. The direction of the symmetry axis can either be given in direct or reciprocal space. A pure mirror operation is performed by an improper rotation by 180°.

## 4.3 Generalised shear operations

A similar operation is required if you want to distort the structure or part of the structure. Such a distortion may occur upon a orthorhombic to monoclinic transformation. If the whole crystal is sheared under this transformation it is, of course, easier to simply change the lattice constants. If, however, only part of the structure is to be deformed to produce for example a fish bone like pattern a local shear operation may be required.

The shear menu in DISCUS allows to perform a shear operation:

$$S = \begin{pmatrix} a_{11} & a_{12} & a_{13} & t_1 \\ a_{21} & a_{22} & a_{23} & t_2 \\ a_{31} & a_{32} & a_{33} & t_3 \\ 0 & 0 & 0 & 1 \end{pmatrix} \quad (4.2)$$

Here the matrix elements may take up any value in order to perform the desired distortion. DISCUS allows several ways to set up the shear operation, see the on-line help for details. You can specify the full matrix, define the Eigenvectors of a shear or define the shear by a plane parallel to which a shear operation is performed.

Additionally DISCUS allows one to select the atom or molecule types to be included in the shear operation using the `sele/dese` and `mse1/mdes` commands.

## 4.4 Unit cell transformations

A common task in crystallography is to transform the unit cell of a crystal, i.e. use an alternative setting for some specific reason. Subsequently one needs to transform the coordinates of the atoms within the crystal to the new set of basis vectors. The `trans` segment of DISCUS gives various options to perform this type of task. The transformation can either be defined as new unit cell in terms of the old unit cell or in terms of the new atom coordinates with respect to the old ones and *vice versa*. Independent of this choice, the origin can be shifted by a user defined amount. DISCUS allows the user to select individual atoms or to transform the complete crystal to the new system. If all atoms in the crystal are transformed to the new base vectors, then the unit cell dimensions and the metric tensor are transformed as well.

Let us assume we have a cubic crystal defined by  $a = b = c = 5\text{\AA}$ . The base vectors for the old system are  $\mathbf{a}$ ,  $\mathbf{b}$ ,  $\mathbf{c}$ . The new set of basis vectors defining the new unit cell shall given by  $\mathbf{a}'$ ,  $\mathbf{b}'$ ,  $\mathbf{c}'$  which are given by the relations  $\mathbf{a}' = \mathbf{a} + \mathbf{b}$ ,  $\mathbf{b}' = \mathbf{a} - \mathbf{b}$  and  $\mathbf{c}' = \mathbf{c}$ . After entering the `trans` segment of the program DISCUS, these definitions could be entered as follows:

```
anew 1.0, 1.0, 0.0
bnew 1.0,-1.0, 0.0
cnew 0.0, 0.0, 1.0
```

Note that the relation between the old and the new system could also be defined with respect to the atom coordinates. All those relationships are calculated by DISCUS and can be displayed using the `show` command. Part of the output for our example can be found below:

```
( a(new) ) = ( 1.00000, 1.00000, .00000 ) ( a(old) )
( b(new) ) = ( 1.00000,-1.00000, .00000 ) * ( b(old) )
( c(new) ) = ( .00000, .00000, 1.00000 ) ( c(old) )

( a(old) ) = ( .50000, .50000, .00000 ) ( a(new) )
( b(old) ) = ( .50000, -.50000, .00000 ) * ( b(new) )
( c(old) ) = ( .00000, .00000, 1.00000 ) ( c(new) )

( x(new) ) = ( .50000, .50000, .00000 ) ( x(old) ) ( .00000)
( y(new) ) = ( .50000, -.50000, .00000 ) * ( y(old) ) + ( .00000)
( z(new) ) = ( .00000, .00000, 1.00000 ) ( z(old) ) ( .00000)

( x(old) ) = ( 1.00000, 1.00000, .00000 ) ( x(new) ) ( .00000)
( y(old) ) = ( 1.00000,-1.00000, .00000 ) * ( y(new) ) + ( .00000)
( z(old) ) = ( .00000, .00000, 1.00000 ) ( z(new) ) ( .00000)
```

This particular example might seem very simple but as soon as e.g. a unit cell transformation in a triclinic system is needed, this part of DISCUS becomes the most popular feature.

## Chapter 5

# Simple crystal modifications

Chapter 2 discussed techniques to create a perfect model crystal from the content of the asymmetric unit. This section will give an overview of simple modifications of single atoms within a model crystal. Section 6 will describe DISCUS tools to modify a complete crystal.

### 5.1 Modifications using variables

Each atom  $\langle i \rangle$  within the model crystal stored by the program DISCUS is associated with a set of variables  $x[\langle i \rangle]$ ,  $y[\langle i \rangle]$  and  $z[\langle i \rangle]$  that describe its position and the variable  $m[\langle i \rangle]$  that contains the atom type (see also section 1.3). In principle every desired defect structure might be realised using these variables and the FORTRAN style loops and conditional statements (see chapter FORTRAN style interpreter in the package manual). One should be aware that this procedure may be very slow for complex problems and large model crystals. Since the variables are addressed using the atom index  $\langle i \rangle$ , a knowledge about the internal storage as described in section 2.3 is very important. The command `trans` in the 'chem' segment of DISCUS allows one to convert between atom index and the corresponding unit cell and site. To illustrate the use of variables, the following type of defect should be constructed using DISCUS. The starting structure is a 10x10x1 unit cell square symmetric crystal with Zr on (0,0,0) and a lattice constant of  $a=5\text{\AA}$ . The defects consist of randomly introduced vacancies on Zr sites and nearest neighbors surrounding the vacancy should be relaxed towards the vacant site. The perfect starting structure and the resulting disordered structure are shown in figure 5.1. Here is the macro used to create the described defects. Again the line numbers are only shown for convenience and not part of the macro itself. To achieve a high degree of flexibility values like crystal size or the number of defects to be created are stored in variables at the beginning of the macro. The desired crystal size is stores in variables  $i[1]$ ,  $i[2]$  and  $i[3]$  (lines 1-3). The variable  $i[4]$  (line 5) gives the number of defects to be created and  $r[1]$  (line 6) specifies the relaxation (here 30 %) of the surrounding neighbors towards the vacant site.

```
1  i[1] = 10
2  i[2] = 10
3  i[3] = 1
4  #
5  i[4] = 6
6  r[1] = 0.30
7  #
```

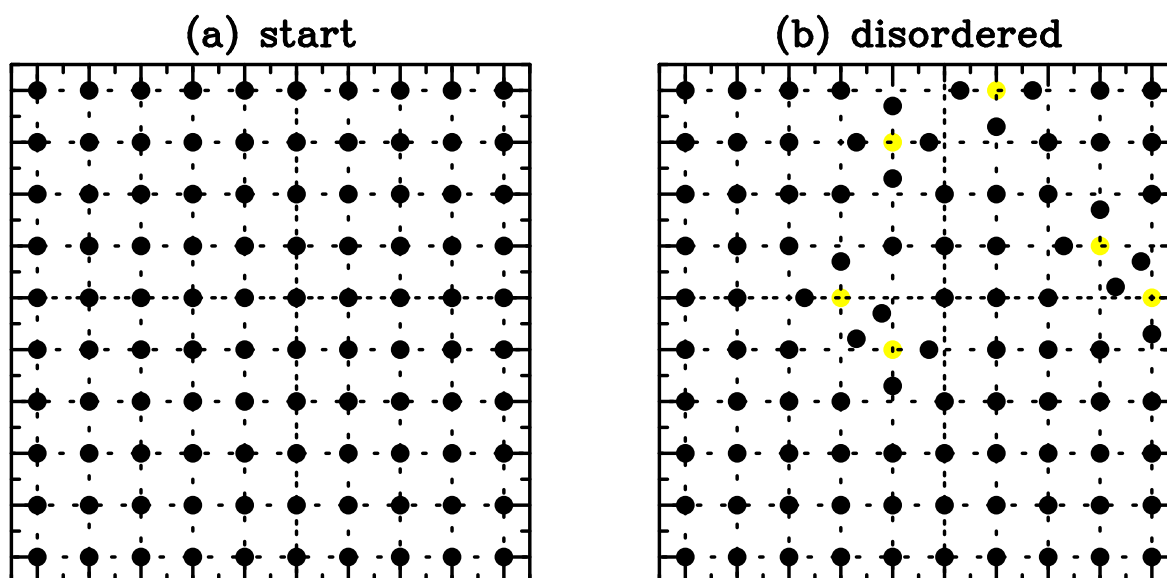


Figure 5.1: Structures created by crystal modification example

Next we read the unit cell from file *cell.cll* and expand it to the desired size (lines 8-9). The macro *plot.mac* (line 11) saves the starting structure suitable for plotting with *KUPLOT* as seen in figure 5.1a).

```

8  read
9  cell cell.cll,i[1],i[2],i[3]
10 #
11 @plot before.plot
12 #

```

The next part (lines 13-15) is needed to switch off periodic crystal boundaries for the command *find* (line 24), otherwise our simple way of relaxing the neighbors would not work.

```

13 chem
14 set mode,quick,noperiodic
15 exit

```

Now the creation of the disordered structure starts with a loop over the number of defects to be created (line 19). Next a crystal site is chosen at random (line 20). Note that the function *ran(0)* produces a random number between 0 and 1 which is multiplied with the number of atoms within the crystal (*n[3]* contains the number of atoms per unit cell, see table 1.2 in section 1.3). If an occupied site was picked (line 22) the atom is removed (line 23). The command *find* (line 24) returns all atoms of the type *Zr* around the position of the selected site within a radius of  $5.5\text{\AA}$ . The atom indices of these nearest neighbors are stored in the variables *env[<i>]* and *env[0]* contains the number of found neighbors. Finally the positions of all neighboring atoms returned by *find* are moved towards the vacant site (lines 27-29).

```

16 #
17 # Loop number of wanted defects
18 #
19 do i[5]=1,i[4]

```

```

20   i[6]=int(ran(0)*i[1]*i[2]*i[3]*n[3])+1
21   #
22   if(m[i[6]].eq.1) then
23     m[i[6]]=0
24     find env,zr,x[i[6]],y[i[6]],z[i[6]],5.5
25   #
26     do i[7]=1,env[0]
27       x[env[i[7]]]=x[env[i[7]]]-r[1]*(x[env[i[7]]]-x[i[6]])
28       y[env[i[7]]]=y[env[i[7]]]-r[1]*(y[env[i[7]]]-y[i[6]])
29       z[env[i[7]]]=z[env[i[7]]]-r[1]*(z[env[i[7]]]-z[i[6]])
30     enddo
31   #
32   endif
33   #
34   enddo
35   #
36   @plot after.plot

```

The resulting structure can be seen in figure 5.1b. The introduced vacancies are shown as yellow circles (or light grey). Since this simple macro does not check whether neighboring atoms are already displaced, the clustered defects in the upper right quadrant of figure 5.1b have a different local environment compared to the isolated defects.

## 5.2 Build in functions

This section will give an overview of those DISCUS functions modifying single atoms or molecules. Some of these functions can be realised using variables as well, others cannot like the command `insert` to insert a new atom in the model crystal. Table 5.1 summarises the available commands. Basically there are three commands to insert a new atom in the structure. The

Command	Description
<code>append</code>	Appends an atom at a given position within the crystal if no other atoms are present in a given distance.
<code>copy</code>	Copies an atom to a different position given absolute or relative to the other position.
<code>insert</code>	Inserts a new atom at the given position without condition.
<code>kick</code>	Inserts a new atom at a given position and removes all other atoms within a given distance from the new atom.
<code>remove</code>	Removes an atom or molecule from the crystal.
<code>switch</code>	Swaps two given atoms or molecules within the crystal.
<code>purge</code>	Removes vacancies (VOID) from the crystal ( <b>use with care</b> )
<code>replace</code>	Replaces one or more atoms or molecules with the given type.

**Table 5.1:** DISCUS commands for single atoms or molecules

command `insert` just creates a new atom at a specified position without taking notice of its environment. Thus the new atom might be on top of an existing one. In order to avoid this problem, DISCUS provides two other commands `append` which will not insert the atom if there are other atoms within a given distance and `kick` which will remove those atoms close by before inserting the new one. The command `copy` allows the user to copy an existing atom



to a new position that can be given in absolute coordinated or relative to the position of the atom to be copied. Note, that these commands are limited to atoms. New molecule types can be created and molecules copied using the generalized symmetry segment of DISCUS described in chapter 4.2.

The commands `remove` and `switch` work for atoms as well as molecules. Not surprisingly, `remove` removes an atom or molecule and `switch` swaps two atoms or molecules with respect to their location within the crystal.

The commands mentioned in this section so far operate only on a single atom or molecule. The command `replace` can replace a single atom or molecule by a given type or replace more until a given concentration is reached. Although this can be easily realised using the FORTRAN interpreter and variables associated with the crystal, the internal function is much faster, especially for large crystal sizes. The command `purge` will remove all vacancies (VOID) within the crystal. Note that if you remove an atom or a molecule by the `remove` command, DISCUS will not actually remove atoms but rather replace their atom type with VOID. Thus when saving the structure a large number of VOID's might appear in the number. The use of `purge` actually removes those VOID atoms. However, the use of the `purge` command is **not recommended** since many DISCUS function require the crystal to be build in a given order, i.e. having the same number of atom sites within every unit cell either occupied by an atom or a VOID.

## 5.3 Inserting extended objects

In order to insert extended objects for small angle scattering or domains 6.3 DISCUS offers an insert menu. In contrast to the insert command for individual atoms, the menu is evoked by the `insert` command followed by the keyword `domain` or `object`. A corresponding version for molecules is under construction.

### 5.3.1 Inserting domains

Within the DISCUS formalism, a domain 6.3 is essentially a molecule which will be interpreted differently. The domain allows you to place an extended defect into a host structure. The `insert domain` menu allows you to specify the chracter and location of the domain to be inserted.

## Chapter 6

# Build in defect models

The following built in defect models modify the whole structure. They offer extended defects that include modification of many atoms throughout the whole crystal. While such modifications could be achieved as well by the user through extended use of the FORTRAN interpreter, this would require extremely lengthy and necessarily slow macros. Defect models built in the source code of the program on the other hand are calculated in much shorter time. The defect models were implemented with as many general features as seemed necessary. They use separate sub menus to define the intended defect structure.

### 6.1 Thermal displacements

The command `therm` can be used to randomly displace all atoms or rigid molecules of the crystal as to be expected by purely random thermal disorder. The direction of the displacement is distributed with a uniform random spherical distribution. The amplitude of the displacement is distributed by a Gaussian distribution. The mean of the distribution is zero, its sigma is calculated from the isotropic thermal coefficient,  $B$ , of each atom as:

$$\langle u^2 \rangle = \frac{B}{8\pi^2} \quad (6.1)$$

After the command `therm` is executed, a summary of theoretical and achieved displacement averages is displayed on the screen. An example is shown below:

Thermal displacements summary :

Atom	Input		Achieved			Maximum displacement		
	B	<u**2>	<ux**2>	<uy**2>	<uz**2>	x	y	z
LA(1)	0.34	0.0043	0.0043	0.0043	0.0044	0.2299	0.2450	0.2487
MN(2)	0.21	0.0027	0.0027	0.0027	0.0026	0.1980	0.1868	0.2184
O(3)	0.50	0.0063	0.0061	0.0063	0.0064	0.2827	0.3518	0.3011
O(4)	0.43	0.0054	0.0054	0.0055	0.0055	0.2745	0.3036	0.2855

The first column lists the name of the atom type followed by the  $B$  and corresponding  $\langle u^2 \rangle$  value. The next three columns show the achieved  $\langle u_i^2 \rangle$  values for the x-, y- and z-direction. The last three columns give the maximum displacements in the three directions that were introduced. The values are given in  $\text{\AA}^2$  and  $\text{\AA}$ .

The parameter `mol` will displace rigid molecules according to the `B` value of the atom at the origin of the molecule. Furthermore the parameter `2d` allows to restrict the displacements to directions with more than one unit cell extension of the crystal. This might be used when working with 2-dimensional model crystals where a displacement in the third direction is not wanted.

## 6.2 Modulations

The given structure can be modulated using the wave segment of the program `DISCUS`. Three different types of waves are available, density waves modulating the occupation of sites within the crystal, displacements waves modulating the position of atoms or molecules and rotational waves which modulate the orientation of molecules by rotating them around a user defined axis. First common features shall be described followed by details about the different wave types in separate sections.

`DISCUS` offers three different wave functions  $w(\mathbf{r})$ , sinusoidal, square and saw tooth defined in equations 6.2, 6.3 and 6.4, respectively. The parameter  $\delta$  for the box shaped wave functions determines its asymmetry. The symmetric case is given by  $\delta = 0.5$ . The computed value of  $w(\mathbf{r})$  modulates density, position or orientation as a function of the position  $\mathbf{r}$  within the crystal depending on the wave type selected.

$$w(\mathbf{r}) = A \cos\left(2\pi \left[ \frac{\mathbf{k}\mathbf{r}}{\lambda} + \psi \right] \right) + A_0 \quad (6.2)$$

$$w(\mathbf{r}) = \begin{cases} A + A_0 & \text{for } \frac{\delta}{2} \leq \left| \frac{\mathbf{k}\mathbf{r}}{\lambda} + \frac{\psi}{360^\circ} \right| < 1 - \frac{\delta}{2} \\ A_0 & \text{otherwise} \end{cases} \quad (6.3)$$

$$w(\mathbf{r}) = A \left[ \frac{\mathbf{k}\mathbf{r}}{\lambda} + \frac{\psi}{360^\circ} \right] + A_0 \quad (6.4)$$

The following list describes the different properties common to all wave types that can be defined by the user.

- *Wave vector  $\mathbf{k}$* : The wave vector  $\mathbf{k}$  defines the *traveling* direction of the wave. The vector is defined using the `vect` command in multiples of the base vectors of the crystal, i.e. in unit cell units.
- *Wave length  $\lambda$* : The wave length  $\lambda$  of the modulation wave is entered using the `len` command. Note that  $\lambda$  must be given in Å.
- *Amplitude  $A$* : The amplitude  $A$  defines the upper limit of displacements (in Å) or rotation angle (in degrees). In case of density waves the probability of replacing an atom or molecule oscillates between an upper and lower limit (see section 6.2). The command to set the amplitude  $A$  is `amp`.
- *Constant offset  $A_0$* : A constant displacement or rotation angle can be added to all selected atoms or molecules using the command `amp0`.
- *Phase  $\psi$* : The phase  $\psi$  of the atom or molecule at (0,0,0) can be altered via the command `phase`.

The argument  $[\mathbf{kr}/\lambda + \psi/360^\circ]$  of the wave functions is limited to the range of -1 to 1 for the box and saw tooth function and to a range of  $\pm 2\pi$  for the sinusoidal wave function. The origin of the vector  $\mathbf{r}$  is the crystals origin. DISCUS allows the user to select which of the atoms in the crystal are to be modified by the wave function.

### Density waves

Displacement waves are selected using the `dens` command. A density wave replaces an existing atom or molecule by another atom or molecule type or alternatively removes the atom or molecule. The direction of the density wave is along the wave vector. A lower and upper probability  $P_{low}$  and  $P_{high}$  for replacing an atom or molecule must be provided using the command `plow` and `phigh`. The probability for the current atom or molecule to remain in the structure oscillates between these two values. The parameters  $A$  and  $A_0$  are calculated from these two probabilities. In case of a sinusoidal wave the parameters  $A$  and  $A_0$  are defined by:

$$\begin{aligned} A &= \frac{1}{2}(P_{high} - P_{low}) \\ A_0 &= \frac{1}{2}(P_{high} + P_{low}) \end{aligned} \quad (6.5)$$

In case of a box or saw tooth shaped wave function, the parameters  $A$  and  $A_0$  are determined by the following equations:

$$\begin{aligned} A &= P_{high} - P_{low} \\ A_0 &= P_{low} \end{aligned} \quad (6.6)$$

A random number is calculated every time an atom is encountered that had been selected. If this random number is less than  $w(\mathbf{r})$ , the amplitude calculated by the wave function, the atom or molecule is retained as was, otherwise it is deleted or replaced by another atom or molecule. Its position is not changed.

### Displacement waves

Displacement waves are selected using the `trans` command. For displacement waves, the wave function  $w(\mathbf{r})$  determines the displacement of an atom or molecule along a given direction of oscillation. The command `osci` defines the oscillation direction in lattice units. DISCUS allows any direction of that vector relative to the propagation direction of the wave. In cases where propagation and oscillation vector are parallel, we speak of longitudinal waves. In cases where propagation and oscillation are normal to each other we speak of a transversal wave. The default displacement mode is set to acoustic, i.e. all atoms are displaced in the same direction. An (admittedly crude) optical mode will displace all atoms that are identified as negative ions in the opposite direction to all others. Negative ions are recognised through their respective name, e.g. `CL1-`. As a side effect, if charged ions are used, the Fourier transform will use the corresponding scattering curve. If this is to be avoided, calculate the desired wave twice, once with positive amplitude and once with negative, while selecting the corresponding atoms.

### Rotational waves

Another feature of DISCUS are rotational waves limited to be used with molecules. Obviously rotating an atom around itself makes no sense. Rotational wave are selected with the `rot` command followed by parameters defining the rotation axis relative to the molecules origin in lattice units. The wave function  $w(\mathbf{r})$  then modulates the rotation angle  $\phi$  around this axis. Additionally an optional offset relative to the molecule origin for the rotation axis can be specified by the `rot` command. An example of such a rotational wave can be seen in figure 6.1.

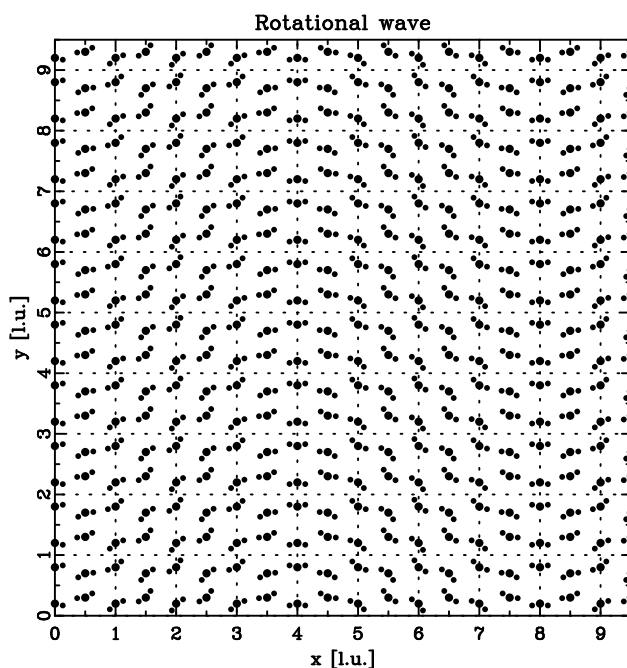


Figure 6.1: Example of a rotational wave

Here a cubic crystal with a size of  $20 \times 20 \times 1$  unit cells ( $a = 10 \text{ \AA}$ ) was used, each containing four  $H_2O$  molecules. The rotation axis is normal to the drawing plane, i.e. in  $z$ -direction. The wave vector is parallel to  $x$  and the wave length  $\lambda$  is eight unit cells ( $80 \text{ \AA}$ ). The amplitude  $A$  was set to  $45.0$ , i.e. the rotation angle  $\phi$  oscillates between  $\pm 45$  degrees.

### 6.3 Domains

The `domain` module of DISCUS has replaced the `microdomain` module. A domain within DISCUS is understood as a place holder for an extended defect. This defect might be as small as an individual atom or be a region that extends over many unit cells.

Note, that in our book (Neder and Proffen, 2007) we present examples using this domain module and discuss domains at length.

As usual, the commands related to this modules are accessed after entering the module via the command `domain`. Let us look at a simple example how to use this module:

```

1  domain
2  rese
3  mode  pseudo
4  input  dom.cube.pseudo
5  assign char,si,sphere
6  assign fuzzy,si,0.5
7  assign cont, si,dm.guest.cube.stru
8  assign shape ,si,1,  1. , 0. , 0. , 0.
9  assign shape ,si,2,  0. , 1. , 0. , 0.
10 assign shape ,si,3,  0. , 0. , 1. , 0.
11 assign orient,si,1,  1. , 0. , 0. , 0.
12 assign orient,si,2,  0. , 1. , 0. , 0.
13 assign orient,si,3,  0. , 0. , 1. , 0.
14 show
15 run
16 exit

```

After resetting the module (line 2), mode `pseudo` is selected. In this mode, a DISCUS structure file is used to define the distribution of domains and each *pseudo atom* in this file is later replaced by the corresponding domain. In our case, the atom Si is replaced by the defined spherical domain (line 5). The boundary is created by removing all atoms that are within 0.5Å of the domain (line 6). Next the domain dimension is defined using the matrix defined in lines 8–10. These values are in terms of unit cells of the host structure. The second matrix (lines 11–13) defines the orientation of the domain. Finally the settings are listed (line 14) and the domain is finally created (line 15). There is an alternative extension of a structure file and the keyword `domain` allows to readily define the domain structure, its shape and orientation (see Neder and Proffen (2007)).

## 6.4 Stacking faults

All crystals whose structures can be described by layers are prone to stacking faults. A stacking fault is any defect that alters the periodic sequence of layers. These defects may be a wrong layer inserted into the sequence, a change of the layer sequence or a different translation between two subsequent layers. These defects may affect the whole crystal or a finite region if e.g. an additional layer is present between an otherwise perfect sequence of layers. DISCUS contains a tool to create layered crystal structures and to introduce stacking faults into these crystals. The crystals are formed in a two step procedure. First, the origin and type of each layer is determined and second, the atoms corresponding to each layer are introduced into the crystal. The user can define each layer type, the translation vectors between consecutive layers and the correlation between neighboring layers. A new feature of the stacking fault part of DISCUS is the addition of rotational disorder for the layer sequences.

The stacking fault sequence is defined by several parameters that can be set in the `stack` sub-level of DISCUS:

- *Type of layers* : The positions of all atoms within a layer are read from a DISCUS type structure file. These layer files have to be created for each layer type involved beforehand using the various DISCUS tools.
- *Translations* : A translation vector between neighboring layers of each type must be provided. Thus N different layer types result in a N\*N matrix of translation vectors. An example for translation vectors in a cubic face centered structure is given in table 6.1.

- *Uncertainties for translation vectors*: In some materials small deviations in the translation vectors might occur. This behavior can be simulated in DISCUS by setting a standard deviation  $\sigma$  to each of the elements of the translation vector matrix. DISCUS will calculate the actual translation vector as sum of the 'ideal' vector plus a Gaussian distributed part defined by the value of  $\sigma$ .
- *Correlations* : A correlation matrix is used to define the probabilities of two layer types to be nearest neighbors. No further correlations are taken into account. To create correlations with *Reichweite* larger than 1, you can build a one dimensional crystal of two or more atom types. The types and positions of these atoms can be interpreted as layer types and origins.
- *Crystal shape* : The resulting crystal can be generated using two different modes: First, the crystal continuously grows in one direction as given by the translation vector(s). Secondly one or two coordinates can be constrained to a finite range, which results in a zig-zag shaped crystal. If any of the parameters is not equal to zero, the corresponding coordinate of the origin is taken modulo this parameter. Note, that DISCUS does not check whether the moduli vectors are translation vectors of the current space group.

Layer type A	Layer type B	Translation vector
A	A	(1, 1, 1)
A	B	$\frac{1}{2}(1, 1, 0)$
A	C	$\frac{1}{2}(1, 0, 1)$
B	A	$\frac{1}{2}(\bar{1}, \bar{1}, 0)$
B	B	(1, 1, 1)
B	C	$\frac{1}{2}(0, 1, 1)$
C	A	$\frac{1}{2}(\bar{1}, 0, \bar{1})$
C	B	$\frac{1}{2}(0, \bar{1}, \bar{1})$
C	C	(1, 1, 1)

**Table 6.1:** Translation vectors for stacking faults in a cubic face centered structure

The command `create` in the `stack` segment of DISCUS creates the list of layer origins and `run` actually generates the corresponding crystal by decorating the origins with the individual layer types. In order to speed up the calculation of the Fourier transform, rather than using the resulting complete structure, the scattering intensity is calculated in the following way. The scattering density  $\rho(\mathbf{r})$  of a layered structure can be expressed as the scattering density of the individual layer types convoluted with the layer origin distribution.

$$\rho(\mathbf{r}) = \sum_{i=1}^{nl} \left\{ \sum_{j=1}^{no} o_{ij}(\mathbf{r}) \right\} \star l_i(\mathbf{r}) \quad (6.7)$$

The outer sum runs over all  $nl$  layer types and the inner sum runs over all origins  $o_{ij}$  of layer type  $i$ . The variable  $l_i$  is the scattering density of layer type  $i$ . Using the convolution theorem,



the Fourier transform of this expression becomes

$$\mathcal{F}\{\rho(\mathbf{r})\} = \sum_{i=1}^{nl} \mathcal{F}\left\{\sum_{j=1}^{no} o_{ij}(\mathbf{r})\right\} \cdot \mathcal{F}\{l_i(\mathbf{r})\} \quad (6.8)$$

Here  $\mathcal{F}$  denotes the Fourier transform. This procedure not only speeds up the calculation but it also allows the usage of much larger crystal sizes since the actual structure does not have to be created in order to calculate the Fourier transform.

## 6.5 Surfaces

An initial model crystal will be a block of  $I \times J \times K$  unit cells. This shape is ideal to avoid finite size effects when you want to calculate diffuse scattering for an infinite large crystal. Nanoparticles or domains are of course explicitly finite objects, whose shape you might want to control. DISCUS offers the `surface` menu to shape the crystal into a finite sized object. The menu replaces the `boundary` command that is still available for backward compatibility.

Within the `surface` menu, the `boundary` command allows you to shape the crystal by a set of lattice planes HKL, by a spherical surface or into the shape of a cylinder.

A plane will remove all atoms to one side of this plane:

```
boundary hkl, 1, 1, -1, 15.0, inside
```

In this example a  $11\bar{1}$  plane is placed at 15.0Å from the origin. All atoms remain that are inside the boundary plane i.e. at the same side of the plane as is the origin 0.0, 0.0, 0.0. As a plane extends to infinity parallel to its surface, DISCUS cannot create an object with an indented surface. Such an object currently must be created in a multiple step fashion. First one would create individual building units and then paste these together via the domain concept.

A spherical object with radius R is create by the command:

```
boundary sphere, 35.0, inside
```

A limitation is that DISCUS places the center of the sphere at the origin. To create an spherical object that is not centered at the origin, simply shift the whole crystal by  $-\vec{R}$ , cut the sphere and shift the object back by  $+\vec{R}$ .

Another limitation is the restriction to spherical symmetry. To create an ellipsoidal object one should shorten the crystal along one axis, cut the sphere and then extend the crystal again:

```
1 radius_a = 20.0
2 radius_c = 40.0
3 @shear.mac radius_a, radius_c
4 surface
5     boundary sphere, 35.0, inside
6 exit
7 @shear.mac radius_c, radius_a
```

We begin by setting two radii, one in the a-b plane and a second along the c-axis. The `shear.mac` macro modifies the length of the crystal along the c-axis by a factor `parameter_1/parameter_2`. After we cut a sphere in lines 4 to 6, the `shear` macro is used again to invert the size change.



```

1  shear
2      matrix 1,0,0, 0,1,0, 0,0,$1/$2      ! define shear matrix row by row
3      origin 0,0,0, crystal                ! Origin is at 0,0,0 compared to crystal
4      sel    all                          ! Select all atom types
5      incl   all                          ! Include all atom numbers
6      show                                       ! Show settings
7      run                                       ! Perform actual calculation
8  exit                                       ! Back to main menu

```

Within the shear menu, the matrix command allows you to define a general affine transformation that is applied to all atoms. The nine parameter to the matrix command are row by row the elements of this matrix. The actual operation applied is:

$$\begin{pmatrix} x' \\ y' \\ z' \end{pmatrix} = \begin{pmatrix} a_{11} & a_{12} & a_{13} \\ a_{21} & a_{22} & a_{23} \\ a_{31} & a_{32} & a_{33} \end{pmatrix} \begin{pmatrix} x \\ y \\ z \end{pmatrix} \quad (6.9)$$

In this example the matrix is a unit matrix except for element 33. Thus the z coordinate of all atoms is multiplied by  $\text{parameter\_1}/\text{parameter\_2} = 20./40.$  and our crystal is compressed along the c axis by a factor of 1/2. After the surface menu the arguments to the shear macro are simply exchanged and the crystal is expanded along the c axis by a factor of 2. Analogous operations could be performed to create a triaxial ellipsoid with arbitrary orientation.

To shape a cylindrical object, use a command like:

```
boundary cylinder, 100., 20.0, inside
```

This will create a cylinder with 100.Å half axis along the c axis and a radius of 20.0Å. The top and bottom planes of the cylinder are parallel to (001). In a monoclinic or triclinic system the plane will not be at 90° to the cylinder axis. To create a cylinder along any other axis, use the symmetry menu to rotate the crystal.

## Chapter 7

# Analysing defect structures

After describing how to create and change structures this chapter will give a summary of DISCUS functions to analyse a given disordered structure. All those commands are accessed via the `chem` sublevel of the program. This segment is entered using the command `chem` and left again by the command `exit`. The command `mode` determines whether the entered commands operate on atoms (`mode atom`) or molecules (`mode mole`). If working with molecules the molecule type number replaces the atom name or number as parameter for the different commands. Note that not all commands are available in molecule mode. Also note, that the DISCUS cookbook (Neder and Proffen, 2007) includes a complete chapter on the analysis of disordered structures with many examples and exercises.

### 7.1 Occupancies

The command `elem` will display the chemistry of the current model crystal. Depending on the selected mode the relative abundance of the present atom or molecule types is written on the screen. The results are saved in the variable `res[i]` (see chapter FORTRAN style interpreter in the package manual). Entry `res[1]` contains the relative amount of atom types `void` and entries `res[1]` the relative amount of atom type 1.

The relative abundances given by the command `elem` are for the complete crystal and contain no further information about its homogeneity. DISCUS allows the user to obtain concentration as well as correlation (see section 7.3) distributions within the crystal using the command `homo`. This is achieved by sampling the crystal using a predefined volume.

### 7.2 Distortions

After analysing concentrations within a crystal, this section will focus on the analysis of distortions. The simplest way is to compute average distances between different atom types in a given range. An example is given below. First the desired range of interatomic distances is limited to values between 4.5Å and 5.5Å (line 1). Next the bond length distribution is calculated using the DISCUS command `blen`. Here distances between *all* atoms will be considered and the resulting distribution is written to the file *chem.2.blen* (line 2).

```

1  set blen,4.5,5.5
2  blen all,all,chem.2.blen

```

Rather than using `all` as parameters for the command `blen`, atom names or numbers could be used to calculate the average bond length only for specify atom types.

The command `blen` averages **all** distances within the given range specified with `set blen`. In cases where more specific information about distances in given directions is needed, the command `disp` must be used. Now it is necessary to enter neighbour definitions which can either be specified by a distance criteria or by individual vectors. In the example below the command `set vec` in line 1 defines vector 1 (first 1) to be from site 1 in one unit cell to site 1 (2nd and 3rd parameter) in the next unit cell in x-direction (1,0,0 - last 3 parameters). Next this vector is used as neighbour definition 1 (line 2). Finally the lattice averages are computed between all present atom types (line 3).

```

1  set vec,1,1,1,1,0,0
2  set neig,vec,1
3  disp all,all

```

### 7.3 Correlations

In this chapter the concept of correlations as a measure for those two-body interactions will be introduced. Although diffuse scattering contains only information about two-body interactions the concepts described here can easily be extended to include multi-site correlations. It should be noted that although these multi-site interactions do not show up in the diffraction pattern directly, they can have a constraining influence on two-body interactions and thus affect the diffraction pattern. However, DISCUS is currently limited to the calculation of two-body interaction averages.

We will talk about atom types in the following section, however, all correlation related commands are available for molecules as well. To work with molecules use the command `mode mole` and specify molecule types rather than atom types or names as parameters for the commands.

#### Occupational correlations

One definition of the correlation coefficient  $c_{ij}$  between a pair of sites  $i$  and  $j$  based on a statistical definition of correlation (Welberry, 1985) is given in equation 7.1.

$$c_{ij} = \frac{P_{ij} - \theta^2}{\theta(1 - \theta)} \quad (7.1)$$

$P_{ij}$  is the joint probability that both sites  $i$  and  $j$  are occupied by the same atom type and  $\theta$  is its overall occupancy. Negative values of  $c_{ij}$  correspond to situations where the two sites  $i$  and  $j$  tend to be occupied by *different* atom types while positive values indicate that sites  $i$  and  $j$  tend to be occupied by the *same* atom type. A correlation value of zero describes a random distribution. The maximum negative value of  $c_{ij}$  for a given concentration  $\theta$  is  $-\theta/(1 - \theta)$  ( $P_{ij} = 0$ ), the maximum positive value is  $+1$  ( $P_{ij} = \theta$ ). Let us look at a simple example:

```

1 read
2 stru chem.1.stru
3 #
4 chem
5 #
6 set mode,quick,periodic
7 #
8 set vec,1,1,1, 1, 0, 0
9 set vec,2,1,1,-1, 0, 0
10 set vec,3,1,1, 0, 1, 0
11 set vec,4,1,1, 0,-1, 0
12 #
13 set vec,5,1,1, 1, 1, 0
14 set vec,6,1,1,-1, 1, 0
15 set vec,7,1,1, 1,-1, 0
16 set vec,8,1,1,-1,-1, 0
17 #
18 set neig,vec,1,2,3,4
19 set neig,add
20 set neig,vec,5,6,7,8
21 #
22 corr occ,zr,void
23 #
24 exit

```

The macro starts with the reading of the disordered structure (lines 1-2). After the chem sub-level is entered (line 4) periodic crystal boundaries are selected (line 6). The parameter `quick` selects a faster neighboring finding algorithm which only works for crystals arranged in the DISCUS storage order (see section 2.3). Note that  $c_{\langle 10 \rangle}$  stands for the nearest neighbor correlations in all four symmetrically equivalent  $\langle 10 \rangle$  directions of the two dimensional cubic test crystal, i.e.  $c_{10}$ ,  $c_{\bar{1}0}$ ,  $c_{01}$  and  $c_{0\bar{1}}$ . All eight neighboring directions for  $c_{\langle 10 \rangle}$  and  $c_{\langle 11 \rangle}$  are defined as vectors 1 to 8 in lines 8 to 16 of the macro file. Next vectors 1 to 4 are grouped as neighboring definition for  $c_{\langle 10 \rangle}$  (line 18) and vectors 5 to 8 for  $c_{\langle 11 \rangle}$  (line 20). The command 'set neig,add' (line 19) stores the current neighboring definition and allows the definition of a new one. Finally the correlations for the defined neighboring directions are calculated (line 22). The screen output looks like this:

```

Calculating correlations
Atom types : A = ZR   and B = VOID

Neig.    AA        AB        BB        # pairs    correlation
-----
1       50.49 %    48.99 %    .51 %    40000      -.3061
2       71.04 %    7.89 %    21.07 %    40000      .7897

```

The program lists the probabilities for AA, AB and BB pairs and the corresponding correlations  $c_{ij}$ . Here the value for  $c_{\langle 10 \rangle}$  is negative, i.e. vacancy neighbors in  $\langle 10 \rangle$  directions tend to be avoided. Neighboring vacancies in  $\langle 11 \rangle$  direction on the other hand are much more likely compared to a random vacancy distribution indicated by the large positive value of  $c_{\langle 11 \rangle}$ .

## Displacement correlations

The correlation coefficient  $c_{ij}$  for displacement correlations between two sites  $i$  and  $j$  is defined as:

$$c_{ij} = \frac{\langle x_i x_j \rangle}{\sqrt{\langle x_i^2 \rangle \langle x_j^2 \rangle}} \quad (7.2)$$

Here  $x_i$  is the displacement of the atom on site  $i$  from the average position in a given direction and  $\langle \cdot \rangle$  stands for the average over the crystal. Again a negative value describes a situation where the pairing of corresponding displacements are less likely than in a crystal with random displacements whereas a positive value indicates a larger than random probability. The definition of neighbors is identical to the example in the previous section. Additionally the command `set neigh,dir` is used to determine the displacement direction to be used. Note that the displacement direction for the two sites  $i$  and  $j$  is not necessarily the same, e.g. one could be interested in the correlation between the x-displacement on one site and the y-displacement on the neighboring site.

### Correlation fields

In the previous sections a correlation  $c_{ij}$  for a given pair of neighboring atoms was computed. An interesting information, however, is how these correlations extend within the crystal. The program DISCUS allows the calculation of correlation fields for occupational and displacement correlations (command: `field`).

## 7.4 Bond valence sums

The concept of *bond valence* has found wide applicability in solid state chemistry (Brown and Altermatt, 1985; Brese and O'Keeffe, 1991). One application is the use of bond-valence sums at atoms as a check on the reliability of a determined local structure. The valence of an atom  $i$  is calculated by the following empirical expression:

$$V_i = \sum_{ij} \exp \left\{ \frac{r_{ij}^0 - d_{ij}}{b} \right\} \quad (7.3)$$

Here  $r_{ij}^0$  and  $b$  are the so-called bond valence parameters,  $d_{ij}$  is the distance between the central atom  $i$  and the neighboring atom  $j$ . The sum goes over all nearest neighbors. The bond valence parameters used by DISCUS were taken from a list compiled by I.D. Brown, McMaster University, Hamilton, Canada from various references. Those parameters can be displayed using the command `show`. An example is given below:

```
discus > show bval,zr4+,o2-
Bond valence parameters ZR4+ - O2- : r0 = 1.9280 b = 0.3700
```

The parameters are specific for a given atom pair, here  $\text{Zr}^{4+}$  and  $\text{O}^{2-}$ . Note that it is required to use the atom names indicating the oxidation state like in the example above. It should also be noted, that bond valence parameters are not available for all pairs of atoms.

## 7.5 Other tools

In this chapter a short summary of functions available in the 'chem' sublevel not discussed previously will be given.

The average structure of a crystal can be calculated using the command `aver`. Occupancies, average positions and standard deviations for these positions are calculated. This command is

not available when working with molecules. The neighborhood of a given atom or molecule can be examined using the commands `neig` and `env`. The command `neig` will use the currently stored neighbor definitions whereas `env` will display **all** atoms found in a given distance from the chosen origin. Finally the conversion between atom index and unit cell / atom site can be made using the equation given in section 2.3 or via the command `trans` in the chem level of DISCUS.

## Chapter 8

# Monte Carlo simulation

In general Monte Carlo (MC) methods can be described as statistical simulation methods involving sequences of random numbers to perform the simulation. The MC algorithm goes back to Metropolis et al. (1953). It works as follows: The total energy of the crystal is expressed as a function of random variables such as site occupancies or displacements from the average structure. The simulation process proceeds as follows: A site within the model crystal is chosen at random and the associated variables are altered by some random amount. The energy difference  $\Delta E$  of the configuration before and after the change is computed. The new configuration is accepted if the transition probability  $P$  given by

$$P = \frac{\exp(-\frac{\Delta E}{kT})}{1 + \exp(-\frac{\Delta E}{kT})} \quad (8.1)$$

is less than a random number  $\eta$ , chosen uniformly in the range  $[0,1]$ .  $T$  is the temperature and  $k$  Boltzmann's constant. Thus the energy of the model crystal is minimised by the MC simulation. Note that a higher temperature  $T$  means that more moves will be accepted that lead to a higher total energy  $E$ . In order to analyse the defect structure of a particular system, the diffraction pattern of the generated defect structure can be calculated and compared to the experimental data. By adjusting near-neighbor interactions defining the energy  $E$  the model can be changed until a match with the experiment is obtained. On the other hand MC simulations can be used to explore the relationship between certain correlations and the corresponding diffraction pattern, e.g. for teaching purposes.

In order to be able to carry out MC simulations, the energy  $E$  of the crystal needs to be defined. The following sections give energies for occupational as well as displacement correlations based on the energy for an Ising model. Again readers are encouraged to have a look at the Discus cookbook (Neder and Proffen, 2007) which contains a detailed discussion, examples and exercises on MC simulations using Discus.

*Note that the current version of DISCUS contains a new rewritten module `mmc` which allows to define more complex energies. The old `mc` module is no longer available.*

## 8.1 Occupational disorder

To represent the distribution of atom or molecule types within a crystal, it is convenient to use Ising spin variables  $\sigma_i = \pm 1$ . Here  $\sigma_i = 1$  means site  $i$  is occupied with type A and  $\sigma_i = -1$  stands for type B being present on site  $i$ . Using these variables, the Hamiltonian (or energy) takes the following form:

$$E_{occ} = \sum_i H\sigma_i + \sum_i \sum_n J_n \sigma_i \sigma_{i-n} + \dots \quad (8.2)$$

The sums are over all sites  $i$  and neighbors  $n$ . The value  $\sigma_{i-n}$  refers to the occupancy (spin) of the neighboring site  $i - n$  of site  $i$ . The quantities  $J_n$  are pair interaction energies corresponding to the neighboring vector defined by  $i$  and  $n$ . Although the Hamiltonian can easily be extended to include multi-site interactions, we will neglect those in this manual. The quantity  $H$  is a single site energy which has the effect of an external field in magnetic Ising models. Here it controls the overall concentration.

The interaction energies  $H$  and  $J_n$  are initially unknown and Discus employs a feedback mechanism to determine their values. This is done in the following way: After a MC cycle, defined as the number of MC steps needed to visit every crystal site once on average, has been carried out, the resulting correlations are computed and compared to the target values defined by the user. If the computed lattice averages are too low, then the corresponding  $H$  and  $J_n$  are decreased by an amount proportional to the difference between calculated and required value and *vice versa*. It should be noted that even the simplest 2D Ising model possesses a phase transition which is important to avoid when using the described feedback mechanism during a MC simulation. Starting with DISCUS version 5.1 a damping algorithm gradually decreases the changes of the interaction energies within the feed back loop. This reduces fluctuations and provides a better convergence towards the desired correlations.

## 8.2 Displacement disorder

The MC simulation technique described above to create structures with certain occupational correlations can quite simply been applied to the case of displacements. Displacement correlations were already discussed in section 7.3. The Ising spin variables  $\sigma_i$  are replaced by continuous variables  $x_i$  describing the displacement of the atom or molecule on site  $i$ . Furthermore we assume that the variable  $x_i$  is Gaussian distributed with mean zero ( $\langle x_i \rangle = 0$ ). Thus the Hamiltonian becomes:

$$E_{dis} = \sum_i \sum_n J_n x_i x_{i-n} \quad (8.3)$$

Here, as before, the first sum is over all sites  $i$  and the second sum is over all neighbors  $n$  of the site  $i$ . In this equation there is no term that depends on  $x_i$  alone, since this would introduce a shift in the average value  $\langle x_i \rangle$ . The MC simulation operates as described before. Note that there are two different modes to model displacements `shift` and `swdisp`. Further details can be found in section 10.4.

Note that the displacements  $x_i$  are taken in the direction defined by `set neig,dir` and the energy defined in equation 8.3 is *blind* to displacement components in other directions. It is



recommended to use the 'swdisp' mode to maintain the overall displacements within the crystal. In cases, however, where the shift mode is used, the generated shifts should be restricted to directions corresponding to the correlations desired in the MC simulation. If e.g. the x-displacements of one atom are correlated with the y-direction of neighbouring atoms, the shifts should be restricted via the command `set move` to the xy-plane.

### 8.3 Creating distortions

In the previous section displacement correlations were introduced, but the average displacements remained constant. The modeling of distortions *via* MC simulations works in a similar way. DISCUS offers two different potentials. The first one uses a Hamiltonian (equation 8.4, where the atoms or molecules move in harmonic potentials (Hooke's law).

$$E_h = \sum_i \sum_n k_n [d_{in} - \tau_{in} d_0]^2 \quad (8.4)$$

The sums are over all sites  $i$  within the crystal and all neighbors  $n$  around site  $i$ . The distance between neighboring atoms or molecules is given by  $d_{in}$  and the average distance is  $d_0$ . The desired distortions are defined by the factor  $\tau_{in}$ . The value  $k_n$  is a force constant for each individual neighbor type  $n$ . The second potential is the Lennard-Jones potential :

$$E_{lj} = \sum_i \sum_{n \neq i} \left[ \frac{A}{d_{in}^M} - \frac{B}{d_{in}^N} \right] \quad (8.5)$$

with

$$A = D \frac{N}{N-M} \tau_{in}^M \text{ and } B = D \frac{M}{N-M} \tau_{in}^N. \quad (8.6)$$

The sums are over all sites  $i$  within the crystal and all neighbours  $n$  around site  $i$ . The values for  $A$  and  $B$  are calculated from the target distance  $\tau_{in}$ , i.e. the distance where Lennard-Jones has its potential minimum, and from the potential depth  $D$  which must be negative.

A standard Lennard-Jones potential is calculated with the repulsive term and  $M=12$  and the attractive term with  $N=6$ . DISCUS allows you to use other exponents as well.

A purely repulsive potential is used to create an even distribution of defects within a crystal:

$$E = -E_\infty + \left( \frac{r - r_{min}}{scale} \right)^M \quad (8.7)$$

Here  $E_{infy}$  is the energy between two atoms at infinite distance. As the MC algorithm looks at changes, its value is not relevant. Atoms at distances shorter than  $r_{min}$  are placed at essentially infinite energy, for longer distances the energy changes according to the exponent  $M$ . A large value of  $M$  will cause a steep descent at short distances, but the descent at longer distances might will be less relevant. To extend the effect of the repulsive potential over long distances, choose a large value of `scale`.

Another potential function is the Buckingham potential:

Angular distortions are realized with a potential energy:

$$E = K (\Theta - \Theta_{ideal})^2 \quad (8.8)$$

Here  $\Theta_{ideal}$  is the indented ideal bond angle.

## 8.4 Working with molecules

Discus allows the use of molecules. The command 'set mole' selects the molecule types to be used for the MC simulation and automatically switches Discus to molecule mode. On the other hand 'set atoms' will return to atoms mode and select the atom types to be used for the MC simulation.

All neighboring definitions work exactly as for atoms. However, the site label used to define neighboring vectors (`set vec`) still refers to the atom site within the crystal regardless of the current working mode. Thus the user has to check which site of the unit cell is occupied by the origin of the selected molecules and define the vectors accordingly. The use of the 'distance' mode to define neighbors works straight forward, however, this mode is much slower compared to using the vector definitions. All operation modes work on rigid molecules. Note, there are currently no MC (or RMC) moves defining rotations of the molecules. Rotations and other symmetry operations can be realised by creating the wanted different orientations of the molecules as different types using the symmetry segment of Discus (see chapter 4.2) and subsequently using the *swap chemistry* mode.

## Chapter 9

# Atomic pair distribution function

The atomic pair distribution function (PDF) can be obtained from powder diffraction data and is a valuable tool for the study of the *local* atomic arrangements in a material. This chapter describes how DISCUS can be used to calculate and refine a PDF. It might be interesting to know about the existence of the program *PDFFIT* (Proffen and Billinge, 1999) which allows the full profile structural least square refinement of a PDF. *PDFFIT* uses a command language and structure file format similar to DISCUS and can be obtained from the same source. The PDF of a given structure can be calculated using the relation:

$$G_c(r) = \frac{1}{r} \sum_i \sum_j \left[ \frac{b_i b_j}{\langle b \rangle^2} \delta(r - r_{ij}) \right] - 4\pi r \rho_0, \quad (9.1)$$

where the sum goes over all pairs of atoms  $i$  and  $j$  within the model crystal separated by  $r_{ij}$ . The scattering power of atom  $i$  is  $b_i$  and  $\langle b \rangle$  is the average scattering power of the sample. In case of neutron scattering  $b_i$  is simply the scattering length, in case of X-rays it is the atomic form factor evaluated at a user define value of  $Q$ . The default value is  $Q = 0$  in which case  $b_i$  is simply the number of electrons of atom  $i$ . Generally there are two different ways to account for displacements (either thermal or static) from the average position. First one can use a large enough model containing the desired displacements and perform an ensemble average. Secondly one can convolute each contribution given by  $\delta(r - r_{ij})$  in (9.1) with a Gaussian accounting for the displacements. Alternatively, the PDF peak width can be approximated by the ensemble average of a distorted structure.

As we have seen before, the experimental PDF is obtained by Fourier transform of the reduced structure factor. However, the accessible range in  $Q$  is limited by  $Q_{max}$ . This can be described by a multiplication of the structure factor up to infinity with a step function cutting off at  $Q = Q_{max}$  resulting in the convolution of the PDF with the Fourier transform  $S(r)$  of the step function. DISCUS models the finite  $Q$ -range by convoluting the model PDF  $G(r)$  with

$$S(r) = \frac{\sin(Q_{max} \cdot r)}{r} \quad (9.2)$$

One last correction applied to the calculated PDF,  $G_c(r)$ , accounts for the limited resolution of the experiment in  $Q$ -space. This leads to a decrease of the PDF peak as a function of  $r$  according to the relation  $\exp(-\sigma_Q^2 r^2 / 2)$ . A detailed discussion of the accuracy of PDF analysis is given in

Toby and Egami (1992). More examples and exercises related to the PDF calculation in DISCUS are in our cookbook (Neder and Proffen, 2007).

## 9.1 Calculating the PDF

The calculated PDF for Nickel is shown in figure 9.1. The PDF was calculated for two different situations: The PDF shown as dotted line was calculated without applying convolution given by  $Q_{max}$ . This is achieved in DISCUS by setting the value of  $Q_{max}$  to zero. The second PDF shown as solid line in figure 9.1 was calculated for  $Q_{max} = 20 \text{ \AA}^{-1}$ . The resulting termination ripples are clearly visible.

```

1 read
2 cell ni.c11,5,5,5
3 #
4 therm
5 #
6 pdf
7   set rang,10.0,0.02
8   set qmax,20.0
9   set qdamp,0.0
10  set rad,xray
11 #
12 calc
13 #
14 save pdf,ni.pdf
15 exit

```

The macro file used to calculate the Nickel PDFs is listed above. First we read the unit cell for Nickel and expand it to a size of 5x5x5 unit cells (lines 1–2). Next we introduce thermal

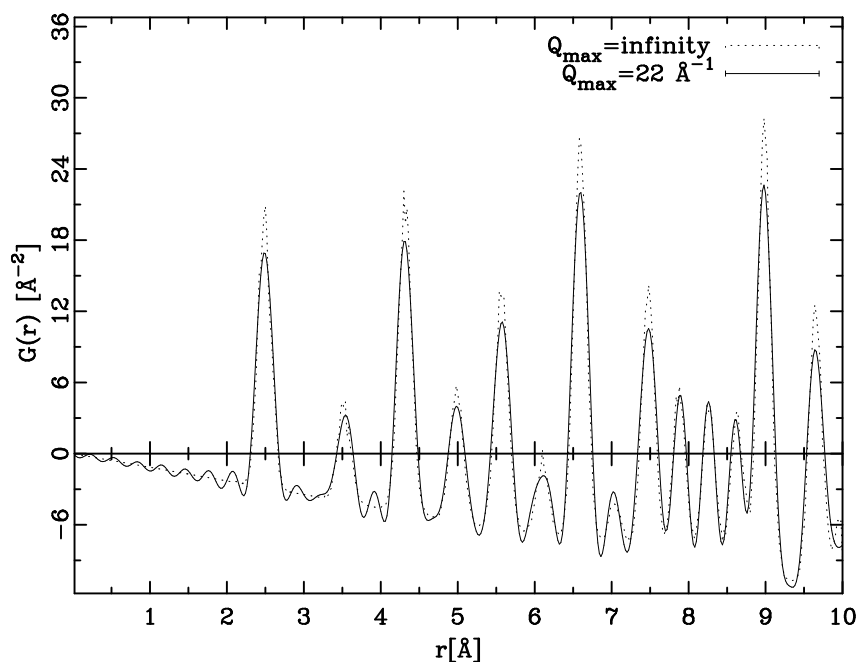


Figure 9.1: Calculated PDFs of *Ni*

vibrations according to the given isotropic Debye-Waller factor (line 4). After entering the PDF sublevel (line 6) we specify the maximum value of  $r$  and grid size  $\Delta r$  (line 7). In our case we calculate up to a value of  $r = 10\text{\AA}$  using a grid of  $\Delta r = 0.02\text{\AA}$ . The next command (line 8) specifies the value of  $Q_{\max}$  to be set to  $20\text{\AA}^{-1}$ . Finally we set dampening term  $\sigma_Q$  to zero (line 9) meaning no Q-resolution correction and select X-ray radiation (line 10). Now we are ready to calculate the PDF, done in line 12. All we have to do now is to save the result to a file (line 14). The result is shown in figure 9.1.

When discussing equation (9.1) used to calculate the PDF from a structural model, we just stated that the sum over  $ij$  goes over *all* pairs of atoms  $i$  and  $j$  within the model crystal. This is perfectly correct to calculate a total PDF. However sometimes it might be desired to calculate just a partial or differential PDF. by selecting the elements of interest.

## 9.2 Refining a PDF

In principle an experimental PDF can be refined based on a structural model in two different ways. A relatively small model can be refined using *PDFFIT*. On the other hand a larger model can be refined using the Reverse Monte Carlo (RMC) algorithm in *DISCUS*. Details about the principle of RMC are discussed in chapter 10 of this manual. The only difference is that rather than refining the scattering intensity directly, the PDF is refined. An example refinement of a Nickel PDF is listed below and is also part of the online tutorial of *DISCUS*.

```
1 read
2 cell ni.stru
3 #
4 pdf
5 data ni.data
6 #
7 set frange,1.5,10.0
8 set qmax,22.0
9 set rad,xray
10 sel ni
11 set mode,shift
12 set move,ni,0.01,0.01,0.01
13 set disp,1
14 set cyc,25
15 show all
16 run
17 save pdf,rmc.pdf
18 save stru,rmc.stru
19 exit
```

In lines 1–2 the starting structure is read. Next the pdf sublevel is entered (line 4). First we read the observed PDF from the file *ni.data*. The maximum  $r$  and  $\Delta r$  which defines the range of the calculated PDF are taken from the data file just read. Then the range in  $r$  that actually should be used for the refinement is set (line 7), here from 1.5 to 10.0 $\text{\AA}$ . In lines 8–9 the value of  $Q_{\max}$  and the radiation used in the experiment is set. Now we enter the RMC related settings (see also 10). We select atoms to be moved (line 10), here Ni. This command should not be confused with *isel* or *jisel* which actually selects the atoms that are included in the calculation of the PDF. The RMC mode is set to shift atoms using a Gaussian distribution with a sigma of 0.01 lattice units ( $\approx 0.035\text{\AA}$ ) in lines 11–12. Finally we set the screen update interval to 1 (line 13) and specify that 25 cycles will be carried out (line 14). Before the refinement is started in line

16, the current settings are displayed using the command `show` (line 15). After the refinement is finished, the resulting PDF (line 17) and structure (line 18) is saved to a file.

## Chapter 10

# Reverse Monte Carlo

This chapter gives a short introduction into the RMC level of DISCUS. DISCUS can be used to either refine the scattering intensities directly or to refine the atomic pair distribution function (PDF) (see chapter 9). A more detailed description of the various commands can be found in the reference manual or the online help function.

### 10.1 Introduction

The **Reverse Monte Carlo** (RMC) method (McGreevy and Pusztai, 1988) is another application of the Monte Carlo algorithm discussed in chapter 8. Here, rather than minimising the total energy of the crystal, the difference between observed and calculated intensity is minimised. Although the method has been around for about 10 years, the method was first applied to *single crystal* diffuse scattering in a neutron diffraction study on ice *Ih* (Nield et al., 1995).

The RMC process starts also with the selection of a random site and changing its variables like occupancy or displacement by a random amount. The scattering intensity is recalculated for the generated move and the goodness-of-fit parameter  $\chi^2$  as given in equation 10.1 is computed.

$$\chi^2 = \sum_{i=1}^N \frac{[I_e(\mathbf{h}_i) - I_c(\mathbf{h}_i)]^2}{\sigma^2} \quad (10.1)$$

The sum is over all measured data points  $\mathbf{h}_i$ ,  $I_e$  stands for the experimental and  $I_c$  for the calculated intensity. The change in the goodness-of-fit is given by  $\Delta\chi^2 = \chi_{old}^2 - \chi_{new}^2$ . Every move which improves the fit to the data ( $\Delta\chi^2 < 0$ ) is accepted. Those moves which worsen the fit ( $\Delta\chi^2 > 0$ ) are accepted with a probability of  $P = \exp(-\Delta\chi^2/2)$ . The parameter  $\sigma$  is assumed to be independent of  $\mathbf{h}$  and is treated as a parameter of the modeling. The value  $\sigma$  can be identified with the temperature  $T$  in the (direct) Monte Carlo method described in chapter 8. How many *bad* moves are accepted depends on the value of  $\sigma$  or  $T$ .

### 10.2 RMC in more detail

For practical use it is necessary to include a scaling factor  $f$  and a background parameter  $b$  in the definition of the goodness-of-fit  $\chi^2$ . A weight  $w(\mathbf{h})$  is included as well. DISCUS allows the

user to choose a particular weighting scheme or to read weights from a separate input file. The definition of  $\chi^2$  used in the program is given in equation 10.2.

$$\chi^2 = \sum_{i=1}^N \frac{w(\mathbf{h}_i)[I_e(\mathbf{h}_i) - (f \cdot I_c(\mathbf{h}_i) + b)]^2}{\sigma^2} \quad (10.2)$$

As in the previous section  $I_e(\mathbf{h}_i)$  stands for the measured intensity at the reciprocal point  $\mathbf{h}_i$ , and  $I_c(\mathbf{h}_i)$  is the calculated intensity in that point. The summation is over all  $N$  experimental data points. The value  $\sigma$  is a parameter of the modeling and controls the fraction of *bad* moves which are accepted. The corresponding parameter in (direct) Monte Carlo simulations is the temperature  $T$ .

Three different ways to calculate the scale  $f$  and background  $b$  are implemented. First the user can define fixed values for both:  $f = f_0, b = b_0$ . Secondly, the background can be set to a fixed value  $b = b_0$  and the scaling factor  $f$  is computed according to equation 10.3.

$$f = \frac{\sum_{i=1}^N w(\mathbf{h}_i)I_e(\mathbf{h}_i)I_c(\mathbf{h}_i) - b_0 \sum_{i=1}^N w(\mathbf{h}_i)I_c(\mathbf{h}_i)}{\sum_{i=1}^N w(\mathbf{h}_i)I_c^2(\mathbf{h}_i)} \quad (10.3)$$

Alternatively both values  $f$  and  $b$  can be refined during the RMC refinement. Equation 10.4 shows the corresponding definitions.

$$\begin{aligned} f &= \frac{\sum_{i=1}^N w(\mathbf{h}_i) \sum_{i=1}^N w(\mathbf{h}_i)I_e(\mathbf{h}_i)I_c(\mathbf{h}_i) - \sum_{i=1}^N w(\mathbf{h}_i)I_e(\mathbf{h}_i) \sum_{i=1}^N w(\mathbf{h}_i)I_c(\mathbf{h}_i)}{\sum_{i=1}^N w(\mathbf{h}_i) \sum_{i=1}^N w(\mathbf{h}_i)I_c^2(\mathbf{h}_i) - \left( \sum_{i=1}^N w(\mathbf{h}_i)I_c(\mathbf{h}_i) \right)^2} \\ b &= \frac{\sum_{i=1}^N w(\mathbf{h}_i)I_e(\mathbf{h}_i) - f \cdot \sum_{i=1}^N w(\mathbf{h}_i)I_c(\mathbf{h}_i)}{\sum_{i=1}^N w(\mathbf{h}_i)} \end{aligned} \quad (10.4)$$

The scaling factor which the program prints on the screen during the RMC refinement is actually (for some yet unknown reason)  $1/f$ . The parameters  $f$  and  $b$  are computed during each RMC cycle and usually have large starting values as long as there are big differences between calculated and observed data. After every RMC move the resulting scattering intensity and the  $\chi^2$  value is calculated. In order to save computing time only the contribution of the modified atoms to the scattering is calculated. The difference  $\Delta\chi^2 = \chi_{old}^2 - \chi_{new}^2$  is taken to decide if the move will be accepted or not. If  $\Delta\chi^2 < 0$  the agreement between calculated and measured data has improved and the move is accepted. Moves which result in a  $\Delta\chi^2 > 0$  are only accepted with a probability of  $P = \exp(-\Delta\chi^2/2)$ . As the value of  $\Delta\chi^2$  is proportional to  $1/\sigma^2$ , the value of  $\sigma$  has an influence on the amount of *bad* moves which will be accepted. Obviously there are two extremes: For very large values of  $\sigma$ , the experimental data are ignored ( $\chi^2 \approx 0$ ) and with very small values of  $\sigma$  the fit ends up in the local minimum closest to the starting point, because there is a negligible probability for *bad* moves. In order to be more independent of the actual



number of data points used, the goodness-of-fit parameter used in the program is given by  $\chi^2 / \sum w(\mathbf{h}_i)$ . The program calculates separate scaling factors and background parameters for every used plane of experimental data. This allows to simultaneously use data measured with X-rays and neutrons, different wavelengths or from different instruments. The goodness-of-fit  $\chi^2$  is displayed as its total value and separate for each data plane.

### 10.3 Setting up a model crystal

The first step in an RMC refinement is the creation of a model crystal of suitable size. In many cases the starting structure will be the (known) average structure for the compound under investigation. Since certain information of the crystal (e.g. symmetry) is used in the RMC segment, it is advisable to **to set the crystal before entering the RMC segment**. Depending on the kind of disorder to be modeled it might be necessary to introduce displacements according to the temperature factors (command 'therm', see section 6.1) or create the needed amount of vacancies before starting the RMC refinement. How to generate a crystal is described in chapter 2, tools to modify the crystal are discussed in chapters 5 and 6.

### 10.4 Operation modes

So far changes to the crystal made during the RMC refinement were simply called 'moves'. These moves can either be a displacement of an atom or the change of the occupancy of an atom site. Because the relative abundance of the elements is not allowed to change during the simulation, the later move is actually made by switching the atoms of two sites within the crystal. The program knows three different operation modes which involve three different kinds of moves shown in figure 10.1. Additionally user defined moves can be included in an external subroutine linked to the program DISCUS. A short description of the different RMC operation modes is given in the list below.

1. *switch chemistry*: This mode (Fig. 10.1a) allows to simulate occupational disorder by selecting two different atoms randomly and switch these two atoms. This operation mode can obviously work only, if at least two different types of atoms are present within the crystal. This mode is selected by the command `set mode, swchem`.
2. *shift atom*: If this mode (Fig. 10.1b) is set, a randomly selected atom is shifted by a random amount in a random direction. The size of the generated shift is chosen uniformly in the interval [-1,1] unit cell. The actual shift applied to the atom is the generated shift multiplied with a user defined factor ('set move,<x>,<y>,<z>'). These factors are also given in unit cell units. The shift mode is selected by with `set mode, shift`.
3. *switch displacements*: This mode (Fig. 10.1c) is selected by the command `set mode, swdisp` by swapping the displacement, i.e. the difference between the average and the actual position, of two randomly selected atoms of the same type and thus the overall average displacements remain constant in contrast the previous mode. The user has to make sure that initial displacements are present in the starting structure (e.g. thermal displacements e.g. by using the command `therm`).

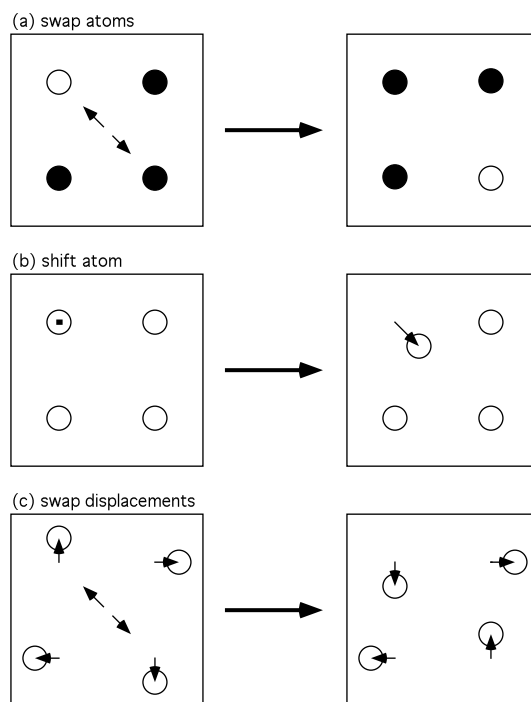


Figure 10.1: RMC operation modes of DISCUS

4. *external*: The program DISCUS allows the user to define more complex RMC moves via an external subroutine. This subroutine is defined in the file *extrmc.f90*. For more details about the construction of such a subroutine, read the commented example in the file *extrmc.f90* which is part of the distribution.

The program allows to select (`sele <typ1>, <typ2>, ...`) and deselect (`dsele <typ1>, <typ2>, ...`) atom types which should be taken into account during the RMC simulation. Alternatively molecules (if present) can be selected using the commands `msele <typ1>, <typ2>, ...` and `mdesele <typ1>, <typ2>, ...`. Here `<typ>` is the corresponding molecule type. All modes listed above can be used for these rigid molecules as well. Rotations and other symmetry operations can be realised by creating the wanted different orientations of the molecules as different types using the symmetry segment of DISCUS (see chapter 4.2) and subsequently using the 'swap chemistry' mode. After every generated move the minimal allowed distances (`set mdis, <atom1>, <atom1>, <dist>`) between all selected atoms are checked and if atoms are too close, the move is rejected.

## 10.5 Running RMC

Finally, the experimental data need to be read before the RMC refinement can start. The file format for the experimental input data is similar to the output formats KUPLOT and PGM for the Fourier transform. It might be necessary to remove Bragg peaks and other unwanted scattering (e.g. powder rings from a sample holder) from the input data set. With the data

command, the method (neutron or X-ray), weighting scheme and the corners of the input data set in reciprocal space are entered. More than one plane of experimental data can be read by repeating the 'data' command. After the data have been read, select the desired RMC mode, select the appropriate atoms and start the refinement with the run command.

```
Gen: 1200 try: 352 acc: (good/bad): 62 / 0 s2x2: 11829643.  
Plane 1: scal: 14.83 / back: 22.23 / s2x2: 11829643.
```

The screen output is updated in a user defined interval, an example output is shown above. The first line gives the number of generated moves ('Gen') and how many of those were actually tested ('try'). The difference is due to selecting atoms that should not take part in the refinement and moves that violate minimal atom distances. The next two numbers give the number of good ( $\Delta\chi^2 < 0$ ) and bad ( $\Delta\chi^2 > 0$ ) moves that have been accepted. The last number is the current value of the overall  $\chi^2$  for all data planes. Additionally the scaling factor  $f$  and the background  $b$  are given for each data plane (one in this example) followed by  $\chi^2$  for the corresponding data plane. The resulting structure as well as the calculated scattering can be saved with the save command.

## Appendix A

# DISCUS commands

### A.1 Summary

```
News          ! Summary of recent changes
addf          ! Add two files together
append       ! Append atom to model crystal
asym         ! Show the contents of the asymmetric unit
boundary     ! Cuts off atoms along an hkl plane
branch       ! Switches to KUPLOT, if within the suite
change       ! Changes the character of an object
chem         ! Enter CHEM level (see chem level for details)
connect      ! Enter CONNECTIVITY level
copy         ! Copy atom or molecule within the crystal
d2r         ! Transform vector from real to reciprocal space
define       ! Define DISCUS specific settings
diff-four    ! Enter DIFFERENCE FOURIER transform level
domain       ! Enter the DOMAIN level
find         ! Find atoms around a given point
four         ! Enter FOURIER transform level
import       ! Imports other formats into DISCUS cells
ins          ! Insert atom or molecule into the crystal
inverse      ! Enter inverse Fourier transform level
kick         ! Insert atom/molecule and possibly remove old atom/mole.
mmc          ! Enter MONTE-CARLO simulations level
output       ! Enter output level (save results)
patterson    ! Enter PATTERSON level
pdf          ! Enter PAIR DISTRIBUTION FUNCTION level
plot         ! Enter PLOT level (export crystal for plotting)
powder       ! Enter POWDER diffraction level
project      ! Calculate projections in real and reciprocal space
property     ! Enter the PROPERTY level
purge        ! Remove voids from crystal (DO NOT USE !)
r2d          ! Transform vector from reciprocal to real space
read         ! Read structure or unit cell from file
remove       ! Remove atom or molecule from crystal
replace      ! Replace atom(s) or molecule(s) with other type
rmc          ! Enter REVERSE-MONTE-CARLO level
save         ! Enter SAVEi level to save the structure
seed         ! Set seed for random number generator
shear        ! Enter the SHEAR level
show         ! Show various information
stack       ! Enter STACKING FAULT level
switch       ! Swap two atoms or molecules within crystal
symm         ! Enter SYMMETRY transformation level
transform    ! Enter unit cell TRANSFORMATION level
```

```
therm      ! Displace atoms or molecules according to B
vprod      ! Calculate vector product in real or reciprocal space
waves      ! Enter WAVE type modulations level
wyckoff     ! Shows symmetry operations of the space group
```

## A.2 News

Here you find a list of recent changes, additions, bug corrections

## A.3 2015February

The speed of the PDF and powder(Debye) calculations has been improved considerably. No side effects on the user.

DISCUS does check if an atom in the unit cell file has a name "void" and it will ignore these atom types when calculating the PDF. This helps to get the proper PDF density and weights for molecules where the first atom is a void at the center of the molecule.

## 2014November

DISCUS will now import CIF files, see ==> import

## 2013September

The chemical short range order in mmc can now sort groups of atom types.

## 2013June

Finally this is it! DISCUS is now a fortran2008 program. Most of the substantial changes should not affect the user (or so we hope). Most of the large arrays are now allocated automatically, as needed, thus the need to compile DISCUS with different size versions should no longer exist.

## Command\_line\_options

The command line may now take the additional option: program -macro <macro\_name> [  
<par1> [ <par2> ...]]

## Electron

DISCUS now offers electron diffraction in the kinematical limit as well. Atom form factors are taken from Table 4.3.2.3 Int. table Vol C (2006) respectively. from Peng, Ren, Dudarev, Whelan Acta Cryst A52 (1996), 257

Electron diffraction works in all menus where you could switch between X-ray and neutrons i.e.: fourier, powder, rmc, pdf

**internal\_storage**

DISCUS now offers an internal storage of crystal structures. Simply start the file name with the string "internal" and DISCUS will write the crystal structure into a dynamically allocated internal data structure, instead of onto the hard disk. If a structure was written into an "internal" file, it can be read as well with ==> 'read'.

**mmc**

The mmc menu can now offer a new "repulsive" energy, and may use the connectivity list to find interacting neighbors

**powder**

In previous versions, DISCUS would write an unevenly spaced powder pattern if you used a Q-axis and a 2Theta output or a 2Theta axis and a Q output. This has now been replaced by an evenly spaced output. If you use a Q-axis with proper Q-range, you also need to specify the 2Theta limits on which you want to write the data. Likewise for a 2Theta axis, you need to specify a Q-range for the output. This change mostly affects the DEBYE equation. Internally, DISCUS always uses an evenly spaced q-grid, as the calculation is faster. With this bug fix, the output will be evenly spaced, for all combinations of axis and output.

**connectivity**

This new menu allows you to define settings for the connectivity list. This connectivity list is used to quickly access neighboring atoms.

**Centering**

In previous versions, DISCUS used the centering generators before the symmetry generators. To be more in line with the International Tables, DISCUS uses now first all symmetry generators, followed by the centering generators. See also the ==> 'define' command to adjust the sequence to your personal taste.

**waves**

The 'mrepl' and 'repl' commands were augmented by a "viceversa" option that will cause atoms/molecules to be replaced by each other rather than only in one way.

**2010sep****property**

A major upgrade of DISCUS adds properties to atoms. These can be N = normal, the atom is a normal atom (instead of a void) M = molecule, the atom is part of a molecule D = domain, the atom is part of a domain O = outside, the atom is outside of the crystal E = external, the atom is close to an external surface I = internal, the atom is close to an internal surface

The new ==> 'property' menu allows global settings, plot uses its local settings. The global settings affect 'replace', 'mmc', 'find env' Commands that add, replace, remove atoms all affect the properties.

### **surface**

The new 'surface' menu covers the old boundary command and sets options to flag the atom property near an external or internal boundary.

### **domain**

A new command 'distance' similar to the corresponding one in the 'surface' menu allows to set distances of atoms to internal boundaries. Atoms closer than these distances are flagged as "close".

### **plot**

The default was changed to plot the entire crystal, even if the 'ext' or 'thick' commands had not been used.

## **2010aug**

### **atom\_coordinates**

The user may now specify the fractional coordinates in a unit cell or a structure file as algebraic expression as in the example:

Si 1.-0.2, 1/3, zpos, bval

Remember, if you use a variable as "zpos" or "bval" in this example, the variable must have been defined up front and been given a sensible value.

To use this feature, it is mandatory, that the fractional coordinates are separated by a comma.

The old style is still possible.

### **unit\_cell\_dimensions**

The user may now specify the unit cell dimensions in a unit cell or a structure file as algebraic expression as in the example:

cell alat, 2\*alat, 1.0+4.0, 90.000, beta, gamma

Remember, if you use a variable as "alat" or "beta" in this example, the variable must have been defined up front and been given a sensible value.

To use this feature, it is mandatory, that the unit cell dimensions are separated by a comma.

The old style is still possible.

## 2008may

### atom\_sequence

In the original source code, the generators for each space group were applied strictly in the sequence as listed in the header of the International Tables.

Thus for centered space groups, each atom was followed immediately by those created by the centering translations. The one exception are those rhombohedral space groups described by hexagonal axes that have a horizontal 2 fold axis and/or a center of inversion. Here the sequence of rhombohedral centering vectors is mixed up.

To correct this, DISCUS now allows the user to choose whether centering generators are applied first or last. The second, new sequence, will correspond more clearly to the printing of symmetry operations for centered space groups as given in the International Tables. Currently the old sequence is still the default, this will be reversed in future releases. Presently the sequence can be defined by ==>

```
define generator, {"center" | "symmetry"}
```

which defines which group of generators comes first.

### molecules

A major issues regarding the assembly of molecules was corrected. In several space groups, molecules were not assembled correctly. See ==> "data keywords molecule" for further details

### symmetry

A new entry in the 'show' menu displays all symmetry operations of the current space group in a variety of formats

### wyckoff

This is a new command at the main menu level that determines the local site symmetry.

### define

This new command defines several settings, just like the ==>"set" command does. The "set" command at the main menu level is reserved for general settings, common to all programs in the DIFFUSE suite. "define" is specific to the individual program.

## 2008april

### formfactors

Within the "fourier" menu, you can define your own values for the atomic form factors as well as anomalous dispersion correction terms. A "reset" was added to switch back to the internal values.



## stacking faults

The new parameter "list" added to the distribution mode, allows you to interpret the coordinates of atoms read from the input file as positions of the respective layers. Previously, the parameter "file" interpreted the atom type only

**boundary**

A cylindrical boundary has been added.

## A.4 Unit cell

!!  
**!! NEW FORMAT OF DATA FILE, STARTING IN DISCUS-3.10 !!**  
 !!!

Don't worry, the old data file format is still supported !

The new format adopts a key word controlled style of the data file. Except for the lines containing the atoms, each line now contains a keyword in the first few characters, followed by the parameters needed for this keyword. The first line MUST contain the keyword "title", the sequence of the other keywords is not relevant. The last keyword MUST be "atoms" which starts the list of atoms. You may include commentaries or empty lines anywhere between the "title" and "atoms" keyword.

The keywords "title", "spcgr", "cell" and "atoms" are required, all other keywords are optional.

### Example

```

title Single atom structure of NA
spcgr Pm-3m
symm 1.00,0.00,0.00,0.123, 0.00,1.0,0.00,0.00, 0.00,0.00,1.00,-0.567
cell 5.0,5.0,5.0,90.,90.,90.
atoms
NA1+ 0.0 0.0 0.0 0.1
NA1+ 1/3, 1-0.5, 0.0, 0.1
NA1+ 1/3, 1-0.5, 0.3, 0.1, 1

```

Further help topics are:

**title**

```
title <title string>
```

The title can be any character string up to 80 characters long

**space**

```
spcgr {<symbol> | no. } [, "2"]
```

The space group symbol is used to generate all atoms within the unit cell. The symbols used should be the Hermann-Mauguin symbols used in Int. Tables Vol.A. A center of inversion

should be given as "-" sign immediately preceding the axis. Lattice types need to be given as capital characters, mirror planes as small characters.

Monoclinic cell choices 2,3 or unique c-axis will be assumed if the corresponding non standard Hermann-Mauguin symbol is used.

Origin choice 2 is indicated by setting the optional second parameter to "2".

The lattice constants are used to distinguish between rhombohedral and hexagonal settings.

The space group symbol is checked for contradictions with the lattice constants. In case of error, the unit cell is not read.

## cell

```
cell <a>, <b>, <c>, <alpha>, <beta>, <gamma>  
cell <a> <b> <c> <alpha> <beta> <gamma>
```

The first form with comma separating the lattice constants is encouraged. Only with this format, can you use a variable instead of fixed unit cell values.

Alternatively, the lattice constants are read in free format <a b c alpha beta gamma> All six constants must be given.

The space group symbol is checked for contradictions with the lattice constants. In case of error, the unit cell is not read.

## atoms

```
<name> <x>, <y>, <z>, <B> [, <propertyflag>]  
<name> <x> <y> <z> <B>
```

If a unit cell is read, only those atoms in the asymmetric unit should be listed. For each of the atoms a line must be given with the name of the atom, the fractional coordinates and the isotropic thermal coefficient B. The property flag is optional. This is, however, automatically written to a saved structure file.

The name (in capital characters) must be leftbound in the first four columns. The fractional coordinates and the B value can be in free format, if the line contains no comma. In this case the property flag must be omitted. Otherwise, the fractional coordinates and the isotropic thermal coefficient B can also be specified as arithmetic expressions.

Optionally charged atoms can be symbolized by e.g. "NA1+". See the help entry 'atom names' for a complete list of atoms for which scattering curves are supplied.

Empty lines in the file are ignored.

## keywords

The keywords "title", "spcgr", "cell", and "atoms" are required, see the help one level up. These further optional keyword allow you to fine tune the unit cell file.

You may include commentaries or empty lines anywhere between the "title" and "atoms" keyword.

Further help topics are:

#

#<comment>

You can provide commands for the data file by including lines that start with a "#". The content of these lines is ignored.

**title**

**title** <titel string>

The string used as title can be any character string up to 80 characters long

**spcgrp**

**spcgr** {<symbol>, <no>} [,"2"]

The "spcgr" keyword defines the space group of the crystal. The space group is used to generate all atoms within the unit cell. The symbols used should be the Hermann-Mauguin symbols used in Int. Tables Vol.A. A center of inversion should be given as "-" sign immediately preceding the axis. Lattice types need to be given as capital characters, mirror planes as small characters.

Monoclinic cell choices 2,3 or unique c-axis will be assumed if the corresponding non standard Hermann-Mauguin symbol is used.

Origin choice 2 is indicated by setting the optional second parameter to "2".

The lattice constants are used to distinguish between rhombohedral and hexagonal settings.

The space group symbol is checked for contradictions with the lattice constants. In case of error, the unit cell is not read.

**scat**

**scat** <name> [,<name>...]

By explicitly listing the atom names on a 'scat' command line, you can define a specific sequence of scattering curves that is independent of the sequence of atoms found in the structure file.

If the structure is read using the 'stru' or 'lcell' commands, DISCUS compares the names and atomic displacement parameters (B-values) to those found on the 'scat' and ==> 'adp' command lines. If an atom name and adp match, the atom is assigned to the scattering curve, otherwise an additional new scattering type is defined. If you read a unit cell with the 'cell' command, the scattering type read from any 'scat' and ==> 'adp' line will define the first scattering types. All atoms that follow will be assigned to new additional scattering types, regardless of their name and adp, even if they match the values from the 'scat' and ==> 'adp' command.

This command is needed if you want to create a list of layer type origins which are read by ==> 'stack'.

You can use more than one 'scat' and ==> 'adp' command in the structure file. The number of atoms on the individual 'scat' and ==> 'adp' lines does not matter, as long as the total number of atom types and adp's is identical. Atom names and adp's are matched pairwise using the

sequence with which they were read from the 'scat' and ==> 'adp' lines. Starting with DISCUS 4.0, the number of atom names on the 'scat' and 'adp' command lines must be identical. Thus, both commands must be present.

## adp

**adp <B-value> [, <B-value>...]**

By explicitly listing the atomic displacement parameters (adp) on a 'adp' command line, you can define a specific sequence of scattering curves that is independent of the sequence of atoms found in the structure file.

If the structure is read using the 'stru' or 'lcell' commands, DISCUS compares the names and atomic displacement parameters (B-values) to those found on the ==> 'scat' and 'adp' command lines. If an atom name and adp match, the atom is assigned to the scattering curve, otherwise an additional new scattering type is defined. If you read a unit cell with the 'cell' command, the scattering type read from any ==> 'scat' and 'adp' line will define the first scattering types. All atoms that follow will be assigned to new additional scattering types, regardless of their name and adp, even if they match the values from the ==> 'scat' and 'adp' command.

This command is needed if you want to create a list of layer type origins which are read by ==> 'stack'.

You can use more than one ==> 'scat' and 'adp' command in the structure file. The number of atoms on the individual ==> 'scat' and 'adp' lines does not matter, as long as the total number of atom types and adp's is identical. Atom names and adp's are matched pairwise using the sequence with which they were read from the ==> 'scat' and 'adp' lines. Starting with DISCUS 4.0, the number of atom names on the 'scat' and 'adp' command lines must be identical. Thus, both commands must be present.

## generator

**generator g11,g12,g13,g14, g21,g22,g23,g24, g31,g32,g33,g34 [, power]**

You can define additional generators through the optional "generator" keyword. These generators act identical to the generators defined through the space group symbol. All previously generated copies of the atom in the asymmetric unit are copied by this generator, and will in turn be copied by any generators following later.

The optional parameter power specifies whether this generator is to be applied only once (default) or whether the generator is to be applied again to the first image. A generator for a three fold axes must be applied once to copy atom x,y,z to -y,x-y,z and then again to copy atom -y,x-y,z to -x+y,-x,z.

Since these additional generators are applied after the space group generators, you can use these generators to create non-standard groups or to create a set of symmetries that does not from a group.

The generator forms the matrix algebra:

$$(g11\ g12\ g13) (x) (g14) (g21\ g22\ g23) * (y) + (g24) (g31\ g32\ g33) (z) (g34)$$

The generators:

```
gene 1,0,0,0.5, 0,1,0,0.5, 0,0,1,0.0, 1
gene 1,0,0,0.5, 0,1,0,0.0, 0,0,1,0.5, 1
```

for example would create the following copies of an atom at 0,0,0:

```
0 ,0 ,0
0.5,0.5,0.0
0.5,0.0,0.5
0.0,0.5,0.5
```

## symmetry

**symmetry s11,s12,s13,s14, s21,s22,s23,s24, s31,s32,s33,s34 [, power]**

You can define additional symmetry operations through the optional "symmetry" keyword. These symmetry operations act different than the generators defined through the space group symbol or listed as additional generators. The symmetry operations copy only those atoms created by the generators. The symmetry operations do not act on copies of the atoms created by previous symmetry operations.

The optional parameter power specifies whether this symmetry operation is to be applied only once (default) or whether the symmetry operation is to be applied again to the first image. A symmetry operation for a three fold axes must be applied once to copy atom  $x,y,z$  to  $-y,x-y,z$  and then again to copy atom  $-y,x-y,z$  to  $-x+y,-x,z$ .

Since these additional symmetry operations are applied after the space group generators, you can use these symmetry operations to create non-standard groups or to create a set of symmetries that does not from a group.

The symmetry operation forms the matrix algebra:

$$(s11\ s12\ s13) \begin{pmatrix} x \\ y \\ z \end{pmatrix} + (s21\ s22\ s23) \begin{pmatrix} y \\ z \end{pmatrix} + (s31\ s32\ s33) \begin{pmatrix} z \end{pmatrix} + (s34)$$

The symmetry operations:

```
symm 1,0,0,0.5, 0,1,0,0.5, 0,0,1,0.0, 1
symm 1,0,0,0.5, 0,1,0,0.0, 0,0,1,0.5, 1
```

for example would create the following copies of an atom at 0,0,0:

```
0 ,0 ,0
0.5,0.5,0.0
0.5,0.0,0.5
```

## cell

```
cell <a>, <b>, <c>, <alpha>, <beta>, <gamma>
cell <a> <b> <c> <alpha> <beta> <gamma>
```

The first form with comma separating the lattice constants is encouraged. Only with this format, can you use a variable or an arithmetic expression instead of fixed unit cell values.

Alternatively, the lattice constants are read in free format `<a b c alpha beta gamma>`. All six constants must be given.

The space group symbol is checked for contradictions with the lattice constants. In case of error, the unit cell is not read.

All six constants must be given.

The space group symbol is checked for contradictions with the lattice constants. In case of error, the unit cell is not read.

## ncells

### ncells nx, ny, nz, ncatoms

This command tells DISCUS how large the crystal is in terms of unit cells along the x,y and z axis. This is useful, if you have stored the crystal created by a previous simulation and want to continue work on this crystal. In order for fast references between atom number and the location of its corresponding unit cell, DISCUS must now know how many unit cells were created by the original ==> 'read/cell' command. <ncatoms> is the number of atoms in each unit cell. You can determine this number by typing the command "eval n[3]".

If you saved the structure, using the keyword controlled format, see help on the 'save' menu, you can tell DISCUS to save the number of unit cells and number of atoms per unit cell values for you.

## atoms

```
<name> <x>, <y>, <z>, <B> [, <propertyflag>]  
<name> <x> <y> <z> <B>
```

If a unit cell is read, only those atoms in the asymmetric unit should be listed. For each of the atoms a line must be given with the name of the atom, the fractional coordinates and the isotropic thermal coefficient B. The property flag is optional. This is, however, automatically written to a saved structure file.

The name (in capital characters) must be leftbound in the first four columns. The fractional coordinates and the B value can be in free format, if the line contains no comma. In this case the property flag must be omitted. Otherwise, the fractional coordinates and the isotropic thermal coefficient B can also be specified as arithmetic expressions.

Optionally charged atoms can be symbolized by e.g. "NA1+". See the help entry 'atom names' for a complete list of atoms for which scattering curves are supplied.

Empty lines in the file are ignored.

## molecule

### molecule [parameter]

This keyword is allowed anywhere between the atoms of the unit cell file. It marks the beginning of a group of atoms that are grouped to form a molecule. Optional parameters are used to specify further details. The first 'molecule' keyword must have no parameters.

Molecules are used in DISCUS for three purposes. Regular molecules define a group of atoms that may be moved around as a group. They do not have to form a molecule in strict chemical sense but are just a group of atoms linked for convenience. Objects define a three dimensional object used to simulate small angle scattering. The pseudoatoms within the "object" describe the position, orientation, and distortion of the "object". Domains are representations of domain origins ==> 'domain'. They allow you to manipulate just the positions, shape, and size of the domains. Once the positions are fixed, you can insert the structure represented by the domain into the host crystal. Valid parameters are:

**end molecule end**

The parameter "end" signals the end of a molecule. All atoms still listed in the unit cell file are treated as individual atoms.

**type molecule type,<type>**

The molecule is of type no. <type>. All molecules of the same type can be handled together by several commands like the symmetry menu.

**generator molecule generator**

This parameter has become obsolete!

DISCUS determines the internal symmetry of the molecule by determining the Wyckof site symmetry of the first atom within the molecule.

The first atom of any molecule that lies on a symmetry element of the space group must be located at the point of highest symmetry of the molecule. If the structure does not have an atom at this site you must include a "void" on this site. This could be the case .e.g. if you have an empty triangle on a threefold axis.

This site symmetry is taken as the internal symmetry of the molecule.

DISCUS compares the lists of atoms created by the space group and by the internal molecule symmetry. Identical sections are linked to one molecule.

Atoms created by other symmetry operations (like lattice centering...) will form a new molecule of the same type.

See the section on site symmetry in the International Tables for further details.

**symmetry molecule symmetry**

This parameter has become obsolete!

**character molecule character,{"atoms"|"cube"|"cylinder"|"sphere"}  
molecule character,{"domain\_cube"|"domain\_cylinder"|"domain\_sphere"}**

A molecule may either represent a list of real atoms, or represent an extended object. The distinction is given through the molecule character. If the character is set to "atoms", the molecules consists of true atoms, and may contain many different atoms. You should calculate the Fourier transformation using the ==> 'fourier' command ==> 'set internal'.

Alternatively, the molecule may represent extended objects. They are considered to consist of continuous matter, without any further internal structure. The boundary is a step function, i.e. outside there is nothing, inside is continuous matter of constant density. Their Fourier transform is calculated using analytical expressions. For each of the available characters, a specific Fourier transformation is calculated. Each object must consist of four pseudo atoms. The first atom defines the atoms, which represent the matter and defines the location of the object within space. The next three atoms represent the orientation and lengths of the axes of the object. If these are not cartesian, the object will not be the ideal object reflected by its name but a distorted object. Each of the axes is the 'half' axis of the object, i.e. half the edge length of a cube, radius and half the length of a cylinder, and the radius of a sphere.



Each axis is represented by the difference vector between the position of the corresponding pseudoatom ((Numbers two to four) and the first atom. This allows you to apply all DISCUS commands to the individual pseudoatoms, thereby changing the size of the object.

The axes refer to the base system you used in the ==> 'cell' keyword of the unit cell file, or set using the ==> 'free' command at the ==> 'read' menu prompt. If you want to create a sphere of 10 Angstroem radius, centered at 0.5,0.5,0.5 in a base system of orthogonal 1 Angstroem long base vectors, the entry should be:

```
cell 1.00, 1.00, 1.00, 90.00, 90.00, 90.00 molecule character,sphere molecule density, 5.0 c 0.500,
0.500, 0.500, 1.45 xaxi 10.500, 0.500, 0.500, 0.0 yaxi 0.500, 10.500, 0.500, 0.0 zaxi 0.500, 0.500,
10.500, 0.0 molecule end
```

Usefull commands to shift objects are the ==> 'symm' menu, to deform an object, use the ==>'shear' menu. New objects can be added to the structure with the ==> 'insert' menu.

### **content molecule content, <type> [,<number>]**

This keyword signals the beginning of a list of content for a molecule. The parameter <type> gives the type number of the molecule. All molecules of identical type can be referred to by a number of DISCUS commands. The optional <number> gives the number of the molecule in the list of molecules. This number is written by the ==> 'save' command. On input the number is ignored! All molecules read via 'molecule content' commands are appended to the list of existing molecule.

The molecule content must be followed by ==> 'molecule atom' commands and terminated by a 'molecule end' command.

### **density molecule density, <value>**

This keyword helps you to create objects of identical volume yet different scattering power. The density is expressed in number of atoms per unit volume within the object.

### **atoms molecule atoms [,<i> ...]**

The atom number <i> is part of the molecule. The number <i> refers to the number of the atom in the current crystal. Up to 20 atoms may be specified on the 'molecule atom' command line.

The molecule content must be terminated by a 'molecule end' command.

## **old\_format**

### **Format of unit cell data file**

The old unit cell data file contains 3 initial lines and one line per atom in the unit cell. All lines are in free format, beginning in the left most column.

```
Line 1: Title up to 80 characters
line 2: Space group symbol [,origin choice number]
line 3: Lattice constants a,b,c,alpha,beta,gamma
```

and for each atom a line with:

```
Name(4 characters) x y z isotropic B
```



Empty lines be in the cell data file will be ignored.

Example

```
Single atom structure of NA
Pm-3m
  5.0 5.0 5.0 90. 90. 90.
NA1+  0.0 0.0 0.0 0.1
```

## A.5 addf

**addf** <resultfile> , <inputfile1> , <scale> , <inputfile2> [, <type>]

Adds two files. The intensity of the second inputfile is multiplied by the scale and added to the intensity read from inputfile 1. Optionally the type of the input files can be given. It defaults to "ni" for files in the standard format. type should be "gnu" for gnuplot files and "1d" for one-dimensional files.

For standard files, the respective headers are compared for consistency. for gnuplot and one dimensional files the h,k,l are compared line by line. If an error is detected, the procedure stops. Thus it is possible for the result files to be incomplete. They are not deleted to allow comparison with the input files.

Any one of the two input files may be identical to the result file, while the two input files must be different files.

## A.6 append

**appe** <name> , <x> , <y> , <z> , <t> , <na> , <ne> , <delx> [, <dely> [, <delz>]]

**appe** <name> , <x> , <y> , <z> , <t> , <na> , <ne> , -<bondlength>

Conditionally inserts an atom of type <name> at the position <x,y,z> in crystal space. The temperature coefficient must be given. 'appe' checks the position <x> , <y> , <z> with respect to all atoms numbered <na> to <ne>. The new atom is NOT inserted, if any of these atoms is within the block <x> +- <delx>; <y> +- <dely>; <z> +- <delz>. If <dely> and/or <delz> are not given, then they default to the value of the last <del.> given on this 'appe' command.

Note, that the <del.> are in fractional coordinates, NOT in Angstrom.

In the alternative form, the eighth's parameter is interpreted as a bond distance. To distinguish the two forms, the bondlength must be less than zero. The new atom is inserted, if no atom is less than <bondlength> Angstrom from the position <x> , <y> , <z>.

If the crystal is empty, i.e. n[1]=0, the 'append' command ignores the parameters <na> , <ne> , <delx> [, <dely> [, <delz>]]. Since there is no atom within the crystal, there is no need for a comparison. Effectively, the 'append' command acts just as an equivalent 'insert' command.

The parameters <na> and <ne> must be in the range 1 to current number of atoms in the crystal i.e. the value of variable "n[1]", and <na> must be less or equal to <ne>. The one exception is <na> = n[1]+1 and <ne> = n[1]. This allows you to start building e.g. a new layer within which you would like to optionally insert atoms without affecting the previous layer. Example i[0] = n[1]+1 do i[1]=1,10 insert al,ran(0),ran(0),0.5, 1.0, i[0],n[1], -2.4 enddo This loop inserts up to 10 Al atoms in the x,y range 0 to 1 at z=0.5 with a minimum distance of 2.4

Angstrom. Since `i[0]`, the value for `<na>` is initialla at `n[1]+1`, all previously inserted atoms are not affected. Since `i[0]` remains fixed within the loop, each new `al` atom affects all `al` atoms previously inserted within this loop.

## A.7 asym

### asym

Shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the variable `m[<index>]`. If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

## A.8 boundary

**boundary "hkl",`h,k,l` [`distance`] [,{"inside"}|{"outside"}]**

**boundary "sphere",`<radius>` [,{"inside"}|{"outside"}]**

**boundary "cylinder",`<radius>`,`<half-height>` [,{"inside"}|{"outside"}]**

You are encouraged to use this command from within the new `==>` 'surface' menu.

This commands removes all atoms that are outside a boundary surrounding the crystal. "hkl" The surface is defined as a plane through the Miller indices `<h>`,`<k>`,`<l>`. The optional `<distance>` determines the distance between 0,0,0 and the boundary surface. Without this parameter, the distance is automatically calculated from the user supplied `<h>`,`<k>`,`<l>` as  $1./dstar(h,k,l)$ . The optional last parameter determines which atoms remain. If the parameter is "inside" or omitted, all atoms remain that are on the same side of the plane as is the origin. If the parameter is "outside", those atoms remain that are on the other side of the plane as the origin. "sphere" The boundary is defined as a sphere at the origin of direct space. The size of the sphere is defined by its radius in Angstroem. The optional last parameter determines which atoms remain. If the parameter is "inside" or omitted, all atoms remain that are inside the sphere. If the parameter is "outside", those atoms remain that are on the outside of the sphere. "cylinder" The boundary is defined as a cylinder of radius `<radius>` and half-height `<height>`, both measures are in Angstroem. The cylinder axis is always along the c-axis. The top and bottom planes are parallel to the (001) plane. If the a- and b-axes are not at 90 degrees to the c-axis, the top will be tilted accordingly. If necessary, transform into an orthogonal cell prior to the application of the boundary command. The optional last parameter determines which atoms remain. If the parameter is "inside" or omitted, all atoms remain that are inside the sphere. If the parameter is "outside", those atoms remain that are on the outside of the sphere.

## A.9 branch

### branch kuplot

Active within the discuss suite only!

Branches to the "kuplot" section.

Within this section any standard KUPLOT command can be given. The behaviour of "kuplot" is essentially the same as in the stand alone version.

## A.10 change

**change "object","character",<number>,{"atom"|"cube"|"cylinder"|"sphere"}**

The 'change' command allows you to change settings and characteristics. So far, it has only been applied to objects.

**change "object","character",<number>,{"atom"|"cube"|"cylinder"|"sphere"}**

This command allows you to change the character of an extended object. The character of the object Number <number> will be changed to one of {"atom"|"cube"|"cylinder"|"sphere"}.

## A.11 chemistry

### chemistry

Enters the 'chem' sublevel. Here all kind of statistics about the model crystal can be obtained like relative amount of elements, distribution of bond-lengths, ...

NOTE: The MC and CHEM sublevel share some variables which define neighbours etc. and settings made in this sublevel might be altered when using the other sublevel. It is always save to repeat settings when entering this sublevel if both levels are used.

Further information is available for commands

### commands

Valid commands at this sub level are:

```
@      ! Executes a macro (see main help level)
=      ! Algebra (see main help level)
angl   ! Calculate average bond angles for given neighbours
aver   ! Calculates average structure and occupancies
bang   ! Calculates distribution of bond angles
blen   ! Calculates distribution of bond lengths
bval   ! Calculates the bond valence sum for given atom
continue! Continue a stopped macro (see main help level)
corr   ! Calculates neighbour frequencies/correlations
disp   ! Calculated average distances for given neighbours
echo   ! Echos a string (see main help level)
elem   ! Show rel. abundances of elements in crystal
env    ! Find neighbours around atom/site or position
eval   ! Evaluate an expression (see main help level)
exit   ! Terminates the chem sublevel.
field  ! Calculates a correlation field
help   ! Gives on-line help for 'chem' (see main help level)
homo   ! Calculate concentration/correlation distributions
```

```

mode      ! Toggles between atoms and molecules mode
neig      ! Show neighbours for given neighbour definition
set       ! Change settings of 'chem' parameters
show      ! Show current parameters
stop      ! Stops execution of a macro (see main help level)
system    ! Executes operating system command (see main help)
trans     ! Transforms between atomindex and unitcell/site
wait      ! Waits for user input (see main help level)

```

## angl

```

angl { <name>|<number>"all" },{<name>|<number>"all" },
{<name>|<number>"all" } [,<file>]

```

This command calculates the average bond angles and sigmas for the specified triplets of atom types for the given neighbours. The neighbour determination parameters are set via the commands -> 'set neig' and -> 'set ang'. The first atom on the 'angl' command forms the center of the bond angle. If an additional parameter <file> is given, all position parameters of the central atom (in lattice units) are written to the corresponding file for possible further analysis or plotting.

## aver

```

aver [ "one" | "ind" ]

```

This command allows to calculate the average structure of the current model crystal. The average position, the standard deviation and the occupancy of every site within the unit cell is given. The average positions and sigmas are stored in the variable array 'res'.

Without parameters or with parameter "one", all atoms on a site are treated as one type and a common average position is calculated. With parameter "ind", the average position is calculated individually for different atom types that may be present on a given site.

```

Warning: This command works only correct, if the structure is given
in the order created by the 'read cell' command. Appended
atoms are simply ignored ! However, using the PURGE command
is not recommended when using this commands !!!!

```

## bang

```

bang { "all" | <name> | <number> },
{ "all" | <name> | <number> },
{ "all" | <name> | <number> }, [<filename>]

```

This command calculates the bond angle distribution between the selected atoms (by <name>, <number> or "all"). The histogram is saved to the file <filename>. The number of points of the histogram is given by 'set bin,<ip>'. The calculation mode can be switched between "quick" and "exact". The mode "exact" calculates the bondangle for EVERY atom pair within the crystal whereas the mode "quick" only looks at neighbouring unit cells. BUT in the "quick" mode errors can occur due to using the "purge" command or appending atoms to the structure. Allowed bondangle rang, bondlength range, calculation mode and sigmas for a neighbour matching criteria are given with the 'set' command.

**blen**

```
blen { "all" | <name> | <number> },
{ "all" | <name> | <number> }, [<filename>]
```

This command calculates the bond length distribution between the selected atoms (by <name>, <number> or "all"). The histogram is saved to the file <filename>. The number of points of the histogram is given by 'set bin,<ip>'. The calculation mode can be switched between "quick" and "exact". The mode "exact" calculates the bondlength for EVERY atom pair within the crystal whereas the mode "quick" only looks at neighbouring unit cells. BUT in the "quick" mode errors can occur due to using the "purge" command or appending atoms to the structure. Allowed bondlength range, calculation mode and sigmas for a neighbour matching criteria are given with the 'set' command. The resulting distances, standard deviations and number of pairs of atoms are stored in the res[i] variables.

**bval**

```
bval "atom",<ia>,<radius>
bval "site",<nx>,<ny>,<nz>,<site>,<radius>
```

This command calculates the bond valence sum for a given atom. The parameters are identical to those of the command -> env. The atom can either be specified by its index ("atom") or unit cell and site number ("site"). The radius specifies the maximum distance for the function to search for neighbours. The radius is given in Angstroems. Note that bond valence parameters are only available for certain pairs of atoms in a specific oxidation state. It is also required to use the corresponding atom names, e.g. Zr4+ rather than Zr in the structure file. The bond valence parameters are taken from a list compiled by I.D. Brown, McMaster University, Canada. The resulting bond valence sum is stored in variable res[1].

**corr**

```
corr "occ", { <name>|<number> }, { <name>|<number> }
corr "disp",{ <name>|<number>|"all" }, { <name>|<number>|"all" }
```

This command starts the calculation of occupational ("occ") or displacement ("dis") correlations within the crystal. The occ. correlations require the name or number of two DIFFERENT atom types to be given. The displacement correlation calculations requires atom names or "all" to use all atom types. The neighbour determination parameters are set via the commands -> 'set neig' and -> 'set vec'. The displacement directions used to calculate disp. correlations are set using the -> 'set dir' command. Correlation values can be :

```
correlation > 0 : preferred same neighbours / displacement
correlation = 0 : random distribution
correlation < 0 : preferred different neighbours / displacements
```

The correlations are calculated according to the following equations:

occupational	displacement
$P(i,j) - \text{theta}^{**2}$	$\langle x(i)x(j) \rangle$
$c(i,j) = \text{-----}$	$c(i,j) = \text{-----}$

```

theta * (1-theta)          sqrt(<x(i)**2><x(j)**2>)

P(i,j): pair prob. sites i,j  x(i): displacement from average pos.
theta : concentration          <> : average

```

The resulting correlations are stored in the variables `res[<i>]` with `<i>` being the number of the corresponding definition for neighbouring distances or vectors. The variable `res[0]` contains as usual the number of variables returned by the command.

## disp

**disp** { <name>|<number>|"all"},{<name>|<number>|"all" } [,<file>]

This command calculates the average distances and sigmas for the specified pairs of atom types for the given neighbours. The neighbour determination parameters are set via the commands `-> 'set neig'` and `-> 'set vec'`. If an additional parameter `<file>` is given, all difference vectors (in lattice units) are written to the corresponding file for possible further analysis or plotting.

## elem

**elem** [ { "on" | "off" } ]

Displays information about elements in the model crystal like relative abundances of the elements. 'void' stands for a vacancy. The results (range 0..1 rather than %) are stored in the variable `res[<i>]`, with `<i>` the number of the atomtyp. The optional parameter "on" or "off" controls if screen output is produced.

## env

```

env "atom",<ia>,<radius>
env "site",<nx>,<ny>,<nz>,<site>,<radius>
env "pos", <x>,<y>,<z>,<radius>

```

The command 'env' allows the user to print out all atoms within a distance of `<radius>` A around the given positions. This position can be given by an atom index `<ia>` ("atom"), a unit cell `<nx>`, `<ny>`, `<nz>` and site `<site>` ("site") or as absolute position `<x>`, `<y>`, `<z>` in lattice units ("pos"). The neighbour determination mode and possible periodic boundaries can be defined using the command `-> 'set mode'`. The atom indices of the neighbouring atoms are stored in the variable `res[i]`. The variable `res[0]` contains the number of neighbours that were found.

## field

**field** {"occ" | "disp"},<a1>,<a2>,<file>,<xmin>,<xmax> [,<ymin>,<ymax>]

This command allows the calculation of a correlation field. The first parameter determines the use of 'occupational' or 'displacement' correlations. The next two parameters specify the atoms to be used. They can be specified as number or name (`-> corr`). The next parameter is the name `<file>` for the output graphics file that can be read by the plot program KUPLOT.

The last parameters define the range in x- and y-direction. If no y-values are given, only a 1-dimensional field is calculated. The values of <xmin>, <xmax>, <ymin> and <ymax> are multiples of the neighbour vectors defined.

The following example will compute the correlation field along <100> up to the 20th neighbour:

```
discus/chem> set vec,1,1, 1,0,0
discus/chem> set neig,vec,1
discus/chem> field zr,void,corr.xy,0,20
```

## homo

**homo "occ",<atom>,<file>**

**homo "cor",{ "occ" | "dis" },<at1>,<at2>,<file>**

This command computes the concentration or correlation distribution within the crystal. The sampling volumes are set using the -> 'set lots' command. The size of the resulting histogram can be altered using the command -> 'set bin'. The following two options are available:

"occ": Here the concentration distribution of the atom <atom> is computed. The resulting histogram is written to the file named <file>.

"cor": Here the correlation distribution for all defined neighbours is computed. The third parameter determines whether occupational or displacement correlations are computed. The next two parameters specified the atom types and the last parameter the filename for the histogram output. Note that \*all\* defined neighbours will be written to the file. To get a specific correlation distribution use a single neighbour definition.

## mode

**mode [ { "atom" | "mole" } ]**

This command determines whether CHEM commands operate on atoms or on molecules. Note that when working with molecules the atom names or numbers in the command descriptions have to be replaced with molecule types. Note that not all commands are available in molecule mode (e.g. 'aver', 'blen').

## neig

**neig <index>,<ic>**

This command determines and displays the neighbouring atoms or molecules around a given atom or molecule given by <index> for the neighbour definition <ic> (-> 'set neig').

The resulting atom or molecule indices of the determined neighbours are stored in the variable res[i] where i is the number of the neighbour. The variable res[0] contains the number of found neighbours.

**set****set** <subcommand>,...

This command sets various parameters within the 'chem' section of the DISCUS program. Valid <subcommands> are:

**commands**

```

"ang"      : defines the neighbors for bond angle calculations
"bang"     : sets range for bond angle calculations
"blen"     : sets range for bond length calculations
"bin"      : sets number of points for bond length histogram
"cryst"    : sets crystal dimensions (if wrong determined by read stru)
"lots"     : sets crystal sampling volumes for 'homo'
"mode"     : sets calculation mode ("quick"/"exact")
"neig"     : defines how correlations are calculated
"vec"      : defines correlation vectors

```

**ang****set "ang",<iv>,<is1>,<is2>,<dx2>,<dy2>,<dz2>,<is3>,<dx3>,<dy3>,<dz3>**

This command is used to define the neighbors for bond angle calculations. To group these neighbor definitions use the ==> 'set neig' command. The first parameter <iv> is the number of the angle to be defined. The variable <is1> gives the number of the crystal site at the center of the angle. Variables <is2> and <is3> give the number of the crystal site at each end of the angle. The values of <dx\*>,<dy\*> and <dz\*> define relative position of the neighbors in the same or adjacent unit cells. Neighbours within one unit cell have <dx\*>,<dy\*> and <dz\*> set to zero. If you want to define angles crossing the unit cell boundaries set <dx\*>,<dy\*> and <dz\*> accordingly.

**bang****set "bang",<min>,<max>**

This command sets the range <min> to <max> of the bond-angle to be calculated and binned to the histogram.

**blen****set "blen",<min>,<max>**

This command sets the range <min> to <max> of the bond-length to be calculated and binned to the histogram.

**bin****set "bin",<number>**

This command sets the number of points for the histograms created by 'chem'.



**cryst**

**set "cryst",<nx>,<ny>,<nz>,<natoms>**

A bug in the current version of DISCUS sometimes causes wrong crystal- dimensions. This command allows to rewrite the values for the crystal- dimension <nx>,<ny>,<nz> and number of atoms in a unit cell <natoms>.

**lots**

**lots "off"**

**lots {"box" | "ellipsoid"},<lx>,<ly>,<lz>,<n>,{ "yes" | "no"},[<fname>]**

This command allows the definition of 'small crystal volumes' to be used to check the homogeneity of the crystal (-> 'homo'). For details about the command and its parameters see the help file entry for -> 'four lots'.

**mode**

**set "mode",{ "quick" | "exact" } [, { "period" | "normal" } ] [,<axes>]**

This command switches the calculation mode between "quick" and "exact". In the "quick" mode only the neighbouring unit cells around a certain atom are taken into account. If the command 'purge' is performed or a structure is read where the atoms are not listed in the DISCUS order this mode might fail, because the calculation of the index of an atom in a neighbouring unit cell can be wrong. In this case the "exact" mode should be used which taken ALL other atoms into account.

The second optional parameter determines is periodic boundaries are used. Note that this only works if the "quick" mode is selected and NO extra atoms were added to the crystal.

Note that this setting does not affect the calculation of correlations defined by vectors using the 'set vec' command.

**neig**

**set "neig",<subcommand>**

This command determines how DISCUS calculates the correlation value using the command 'corr'. These settings might be overwritten when using the MC sublevel. The following sub-commands are valid:

**commands**

```
"ang" : Defines neighbours via given angles
"add" : Adds current definition to list of correlations
"dis" : Defines neighbours via distance criteria
"dir" : Defines directions for displacement correlation calculations
"rese" : Resets list of defined neighbours
"vec" : Defines neighbours via given vectors
```

```
ang set "neig","ang",<iv1> [,<iv2>,..]
set "neig","ang",<iv1> ,<iv2>,"range"
```

"Ang" results in DISCUS using correlation angles <iv> to be used for the calculation. The parameters are the numbers <iv> of the correlation angles defined with 'set ang'. If the second command form is given, the correlation angles in the range <iv1> to <iv2> will be used for the calculation.

```
add set "neig","add"
```

"Add" adds the current neighbour definition to the list of definitions executed later by the -> 'corr' command. Use 'show' to get a list of defined correlation vectors/distances. DISCUS saves actually a copy so that no "add" is needed after the last definition is entered.

```
dis set "neig","dis",<u>,<v>,<w>,<fsig> [, <wsig>] [, {"sym" | "nosym"}]
```

"Dis" causes DISCUS to determine the neighbours used for the calculation of the correlations using the given distance and direction (optional). Those neighbours having a distance corresponding to the vector <u>,<v>,<w> given in unit cell units are taken into account for the calculation. The parameter <fsig> sets the allowed difference of the observed distance to the theoretical value ( $\pm <fsig>/2$ ). The next optional parameter gives the allowed deviation from the neighbouring direction given by <u>,<v>,<w>. The last parameter determines if symmetrically equivalent directions to <u>,<v>,<w> should be used as well. If this parameter is omitted ALL neighbours with the correct distance are valid.

```
dir set "neig","dir",<x1>,<y1>,<z1> [,<x2>,<y2>,<z2>]
set "neig","dir","all"
```

This command sets the displacement directions to be used for the calculation of displacement correlations. The parameter "all" will calculate the displacement correlations for displacements in all directions, i.e. the resulting correlation gives the correlation of displacements of two neighbouring atoms regardless of the direction of the displacement itself. One can also give specific directions, e.g. if one wants to determine the correlations of displacements in z-direction of site i with the ones in x direction of site j one would enter the command 'set dir,0,0,1,1,0,0'. Note that the neighbouring vector between sites i and j is set by the command 'set neig'. If <x2>,<y2>,<z2> are not given their values are set to those of <x1>,<y1>,<z1>.

```
rese set "neig","rese"
```

"Rese" resets the list of correlation vector/distances definitions.

```
vec set "neig","vec",<iv1> [,<iv2>,..]
set "neig","ang",<iv1> ,<iv2>,"range"
```

"Vec" results in DISCUS using correlation vectors <iv> to be used for the calculation. The parameters are the numbers <iv> of the correlation vectors defined with 'set vec'. If the second command form is given, the correlation vectors in the range <iv1> to <iv2> will be used for the calculation.

**vec**

**set "vec",<iv>,<is1>,<is2>,<dx>,<dy>,<dz>**

This command is used to define correlation vectors that might be used to calculate correlations within the crystal (-> 'set neig'). The first parameter <iv> is the number of the vector to be defined. The variables <is1> and <is2> give the number of the crystal site at each end of the vector. The values of <dx>,<dy> and <dz> define the vector in unit cells. Correlation vectors within one unit cell have <dx>,<dy> and <dz> set to zero. If you want to define correlation vectors crossing the unit cell boundaries set <dx>,<dy> and <dz> accordingly.

**show****show**

This command lists the current settings and correlation vector definitions on the screen.

**trans**

**trans <atomindex> or**

**trans <x>,<y>,<z>,<site>**

This command allows to transform between a given index of the storage array <atomindex> and the corresponding unitcell <x>,<y>,<z> and site <site>. The parameter not given is calculated ...

Trans stores the results in the "res[]" variable: res[1] : <atomindex> res[2] : <x> res[3] : <y> res[4] : <z> res[5] : <site>

The command is identical to the ==> 'index2cell' command in the main menu.

## A.12 connectivity

Switches to the connectivity menu. At this menu all parameters related to interatomic distances and the connectivity list are defined.

### Commands

```
add      ! Adds a new definition
create   ! Creates the actual connectivity list for each atom
delete   ! Deletes the connectivity list
remove   ! Removes an individual connectivity definition
reset    ! Resets to program start, removes all def.s and the connectivity list
set      ! Sets an individual definition
show     ! Shows the current definitions
```

**add**

**add <central\_atom\_type>,<neigh types>,<rmin>,<rmax>,<def\_name>**

Add a new connectivity definition. For the atom type <central\_atom\_type> a new definition is added. Atoms of types <neigh types> that are in a distance shell from <rmin> to <rmax>

are entered into this definition. The last parameter is a name for this definition. This command just defines the settings of this connectivity. The actual list is created via ==> 'create'.

### **create**

Creates the actual connectivity list. The program performs a loop over all atoms and searches for neighbors that are valid under any of the connectivity definitions. The valid neighbors are recorded into the connectivity list

Warning, keep in mind that ==> 'property' settings will affect the search for neighbors as well. Any old connectivity list is replaced by a new 'create' command.

### **delete**

Deletes the current connectivity list.

### **remove**

**remove** <central\_atom\_type>, {<def.no.>| <def\_name>}

Removes the definition number <def.no.> or name <def\_name> for central atom type <central\_atom\_type>. If further definitions exist for this atom type, they are shifted forward, and their definition no. is decremented.

### **reset**

Resets the connectivity to program start conditions. All definitions are removed and the connectivity list is deleted.

### **set**

**set** <central\_atom\_type>, {<def.no.>| <def\_name>}, <neigh types>, <rmin>, <rmax>

This command allows you to replace the settings for the definition number <def.no.> or name <def\_name> for the central atom type <central\_atom\_type>. The definition must have been created by a previous 'add' command.

### **show**

#### **show**

**show** "connectivity", <atom\_no.>, <def.no.>

Without arguments, the 'show' command shows a list of all connectivity definitions that have been added.

With the fixed first parameter "connectivity", the show command will show the list of neighbors around atom no. <atom\_no.>, which have been found using definition number <def.no.> for the scattering type of atom no. <atom\_no.> The list of atom numbers is copied into the result variable "res".

### A.13 copy

**copy** <s> , <n> , <x> , <y> , <z>

Copies an atom. If <s> is "a" the atom number <n> is copied to absolute positions <x>,<y>,<z> in crystal space, if <s> is "r", a relative motion copies the atom number <n> to the sum of present position and vector <x>,<y>,<z>.

### A.14 d2r

**d2r** <u>,<v>,<w>

This command transforms direct space vector <uvw> into its corresponding reciprocal space coordinates. See "r2d" for the reverse transformation. The resulting vector is stored in "res[1]", "res[2]", "res[3]".

The parallel, equivalent vector, normalized to 1/(length of [u,v,w]) is stored in "res[4]", "res[5]", "res[6]"

### A.15 define

**define generator**, {"center" | "symmetry"}

The define commands sets a variety of parameters that are specific to the special program, here DISCUS. See the ==> "set" command at the "Command Language" help entry for information of general settings common to all programs within the DIFFUSE suite.

**define generator**, {"center" | "symmetry"}

In the original program development, all generators of a given space group were applied strictly in the sequence as printed in the header of the International Tables. Specifically, the centering generators were applied prior to all other generators. Each atom is immediately followed by the corresponding atom to which the centering vector has been added.

In R-centered space groups, a horizontal 2-fold axis, or a center of inversion turns the centering vector upside down, effectively converting it into the other centering vector. Thus in these space groups, the sequence of some atoms is not exactly that as printed in the Tables.

To adjust this, DISCUS now offers a new mode, in which the centering generators are applied after all other generators. Under this new mode, all atoms are in the sequence as printed in the International Tables. For centered space groups, this block of atom is followed by a corresponding block of atoms that is simply translated by the centering vector.

define generator,symmetry turns on the sequence in which symmetry generators are applied first define generator,center turns on the sequence in which centering generators are applied first

Currently the "center" mode is the default value, this will be changed in future releases!

### A.16 diff-four

**diff**

Switches to the difference Fourier level of DISCUS. All input parameters are identical to "inverse" see the help to "inverse" for further details.

The difference Fourier can only be calculated from a "shelxl" type file. The program calculates (Fobs-Fcalc) and then calculates the corresponding inverse Fourier.

## A.17 find

**find "env",{"all"}{<name>|<number>} [,...],x,y,z,min\_radius,max\_radius**

**find "menv",{"all"}{<name>|<number>} [,...],x,y,z,min\_radius,max\_radius**

This command serves to find something. The first parameter defines what DISCUS shall find. "env"

DISCUS finds the environment around a given position. This environment consists of all atoms that are closer than a given radius. Only those atoms are included that fulfill the global property selection rules ==> 'property'.

"menv"

DISCUS finds the environment around a given position. It consists of all molecules that are closer than the given distance.

The last two parameters define the minimum and maximum radii of the shell within DISCUS will search for neighbours. The three parameters immediately prior to the radii define the central position whose environment is to be found. The second to last but five parameter(s) define which of the surrounding atoms are to be included in the environment. If the second parameter is "all" all atoms are included in the environment, else only those atoms specified by <name> or <number> of the scattering curve.

The number of neighbouring atoms is stored in "env[0]", the indices are stored in the next <env[0]> elements of "env". The resulting bond lengths are stored in the first <env[0]> elements of "res". "res[0]" is set to the same value as "env[0]". Thus, for example, it is possible to search selectively for the next oxygen atoms, while ignoring all other atoms.

Examples

```
find all,0,0,0, 0.1,3.05
```

Finds the environment consisting of all atoms at distances larger or equal to 0.1 Angstrom up to 3.05 Angstrom from the origin.

```
find env,o,n, 1.25,-5.25,3.34, 2.0,2.9
```

Finds all oxygen and nitrogen atoms that are between 2.0 and 2.9 Angstrom from position 1.25,-5.25,3.34

## A.18 fourier

**fourier**

This sublevel of DISCUS calculates the Fourier transform of the given crystal structure. The algorithm to speed up the explicit Fourier is based on the program 'diffuse' by B.D. Butler. See also: B.D. Butler & T.R. Welberry, (1992). J. Appl. Cryst. 25, 391-399.

The program allows to calculate X-ray and neutron scattering including isotropic temperature factors and anomalous scattering. The program also allows to subtract the structure factor of the average structure from the calculated diffraction pattern.

You may calculate a single line through reciprocal space, a layer or a three dimensional volume. Further help topics are:

## commands

Valid commands at this level are:

```
@      ! execute a macro file (see main help level)
=      ! assigns the value to a variable (see main help level)
abs     ! defines the coordinate of the abscissa
delf    ! defines the anomalous scattering corrections for an element
disp    ! switches anomalous dispersion on/off
calc    ! calculates at a single point in reciprocal space
continue! Continue a stopped macro (see main help level)
echo    ! echo a string (see main help level)
electron! switches to electron calculation
eval    ! Evaluates an expression for interactive check
exit    ! terminates Fourier level
help    ! help to Fourier commands (see main help level)
layer   ! defines the layer in reciprocal space to be calculated
ll      ! defines the lower left corner in reciprocal space
lots    ! defines the crystal volume to be used for the Fourier transform
lr      ! defines the lower right corner in reciprocal space
nabs    ! defines the number of points along the abscissa
neut    ! switches to neutron diffraction
nord    ! defines the number of points along the ordinate
ntop    ! defines the number of points along the top axis
ord     ! defines the coordinate of the ordinate
run     ! starts the calculation
scat    ! defines the scattering factor for an element
set     ! sets Fourier mode
show    ! shows the current settings for the Fourier transform
stop    ! Stops execution of a macro (see main help level)
system  ! Executes operating system command (see main help)
temp    ! defines whether temperature coefficients are to be used
tl      ! defines the top left corner in reciprocal space
top     ! defines the coordinate of the top axis
ul      ! defines the upper left corner in reciprocal space
wait    ! Waits for user input (see main help level)
wvle    ! Defines the wave length to be used
xray    ! switches to X-ray calculation
```

## abs

**abs <switch>**

Defines which the component of the vector parallel to the abscissa is written to the output file.

## calc

**calc <h>,<k>,<l>**

Calculates the structure factor at a single point h,k,l. The real value is stored in res[1], the imaginary value in res[2]. The corresponding value normalized to one unit cell are stored in res[3] and res[4], respectively. res[0] is set to 4 to indicate that 4 values may be found in res.

**delf**

```
delf { <name> | <number> } , <f'> , <f''>
delf { <name> | <number> | "all" } , "internal"
```

Defines for the element <name> or the scattering curve number <number> the real <f'> and the imaginary <f''> correction terms for the scattering factors. DISCUS calculates the effective atomic form factor as:  $f + f' + i*f''$ . This means that you should enter the value of  $f'$  as a negative number and  $f''$  as a positive number.

The correction terms are used only, if the dispersion is set to anomalous ==> 'disp'.

```
delf { <name> | <number> | "all" } , "internal"
```

The second command form allows you to set the values back to the internally tabulated values. Keep in mind that these only work with wavelengths specified by their name as in ==> wvle moal'. This command just sets the flag for the element. The values are actually not changed, until the anomalous dispersion is set on ==> 'disp'.

**disp**

```
disp [<switch>]
```

switches anomalous dispersion on /off. If <switch> is "anom" anomalous dispersion will be calculated, if the switch is omitted, no anomalous dispersion is calculated.

**electron**

```
electron
```

switches to electron calculation

**layer**

```
layer <e11,e12,e13, e21,e22,e23, e31,e32,e33, inc1,inc2>
```

sets the lower left, lower right and upper left corner of the Fourier plane to be calculated and sets the number of data points along the horizontal and the vertical

**ll**

```
ll <e11> , <e12> , <e13>
```

Sets the lower left corner of the Fourier plane to be calculated.

**lots**

```
lots "off"
```

```
lots {"box" | "ellipsoid"},<lx>,<ly>,<lz>,<n>,{"yes" | "no"}
```

The program allows to Fourier transform large crystals by averaging the scattering from smaller volumes selected at random. This allows to avoid very long correlations which might occur if the complete crystal scatters in phase.



If this option is not wanted and the complete crystal should be transformed at once, this option can be turned off by the command 'lots off'. Otherwise the following parameters must be specified: First the lot shape. The choices are "box" or "ellipsoid". Usually the first letter of the parameter is sufficient. The following parameters <lx>, <ly> and <lz> describe the size of the subvolumes in unit cells. The parameter <n> specifies the number of such volumes to be averaged. The last parameter can be "yes" or "no" and selects if periodic boundary conditions should be applied.

The lot size should be slightly larger than the longest correlations present in the model crystal studied. The number of lots to be averaged should be set to a value that complete crystal will be covered once on average.

**WARNING:**

If the command 'purge' is performed or a structure is read where the atoms are not listed in the DISCUS order this command should be set to 'lots off'.

**lr**

**lr** <e21> , <e22> , <e23>

Sets the lower right corner of the Fourier plane to be calculated.

**nabs**

**nabs** <inc1>

Sets the number of data points that will be calculated along the abscissa.

**neut**

**neut**

Switches to neutron diffraction

**nord**

**nord** <inc2>

Sets the number of data points that will be calculated along the ordinate.

**ntop**

**ntop** <inc2>

Sets the number of data points that will be calculated along the top axis.

**ord**

**ord** <switch>

Defines which the component of the vector parallel to the ordinate is written to the output file.

**run****run**

starts the Fourier calculation. The user is given a rough estimate of the time it will take.

**scat**

```
scat { <name> | <number> },<a1>,<b1>,<a2>,<b2>,<a3>,<b3>,<a4>,<b4>,<c>
```

```
scat { <name> | <number> | "all" }, <name>
```

```
scat { <name> | <number> | "all" }, "internal"
```

The first command form defines for the element <name> or the scattering curve number <number> a new scattering factor in the exponential form. The scattering factor is calculated as:  $f(h) = a(i) * e^{(b(i)/4*h^2)} + c(i)$  For neutron scattering lengths, set a(i) and b(i) to zero.

```
scat { <name> | <number> | "all" }, <name>
```

The second form of the command allows you to set the scattering factor of two elements to identical values. This comes in handy when you would like to use elements like O1, O2 etc for different oxygen species, or for dummy atom names like xxxx. The command scat O1,O would set the scattering factor of O1 to those values of oxygen. The second parameter must be a valid atom name. See the entry "atom names" for a list of valid names.

```
scat { <name> | <number> | "all" }, "internal"
```

The third form finally allows you to set the scattering factor back to the internally tabulated values.

**set**

```
set "aver" [, <per>]
```

```
set {"external"|"internal"}
```

```
set "aver" [, <per>]
```

The Fourier level of DISCUS allows to subtract the average structure factor <F> by setting the parameter <per> to a value not equals zero. The value of <per> sets the percentage of the crystal to be used to calculate <F>.

The average is calculated by putting <per> % of the crystals atoms into a single unit cell which is Fourier transformed and multiplied with the interference function of the lot shape.

**WARNING:**

If the command 'purge' is performed or a structure is read where the atoms are not listed in the DISCUS order the average structure will not be calculated correctly !

```
set {"external"|"internal"}
```

Usually DISCUS will assume that all atoms in the crystal represent real atoms. In this case the internally tabulated atomic form factors are used to calculate the Fourier transformation. You can explicitly set this mode by typing "set internal". As of version 3.4.2 DISCUS can handle extended objects such as cubes, cylinders, and spheres. Their Fourier transform is calculated using the appropriate analytical expression. In order to activate this state, you must set the Fourier to "external".

**show****show**

Shows the current settings for the Fourier transform file

**temp****temp** <switch>

sets whether temperature coefficients are to be used or not. Valid values for <switch> are "use" and "ignore". The first two letters of the switch are sufficient.

**tl****tl** <e41> , <e42> , <e43>

Sets the top left corner of the Fourier volume to be calculated.

**wvle****top****top** <switch>

Defines which the component of the vector parallel to the top axis is written to the output file.

**ul****ul** <e31> , <e32> , <e33>

Sets the upper left corner of the Fourier plane to be calculated.

**wvle****wvle** { <name> | <value> }

Defines the wave length to be used. Anomalous dispersion will only be calculated if the wave length is given as symbol.

**xray****xray**

Switches to X-ray calculation

## A.19 functions

The following DISCUS specific functions exist. For a listing of general intrinsic functions see help entry 'functions' in the 'command language' section of the online help.

```

bang(u1,u2,u3,v1,v2,v3[,w1,w2,w3])
! Returns the bond angle at the site v
! If only vectors u and v are given, the
! angle between u and v is returned.

bang(atom,i1,i2[,i3])
! Returns the bond angle at atom i2.
! If only atoms i1 and i2 are given, the
! angle at the origin between the vectors
! to atoms i1 and i2 is calculated.

bang(envi,i1,i2)
! Returns the bond angle in the center of
! the last find env command. The angle
! between the vectors from the center to
! entries i1 and i2 in the environment is
! calculated.

blen(u1,u2,u3[,v1,v2,v3])
! Returns the length of vector v-u.
! Vector v defaults to zero.

blen(atom,i1,i2)
! Returns the length of the vector
! between atoms i1 and i2

blen(envi,i1)
! Returns the distance from the center of
! the last environment to atom i1 in the
! environment

dstar(h1,h2,h2[,k1,k2,k3])
! Returns the length of reciprocal
! vector k-h. Vector k defaults to zero.

mol_test(i1)
! Returns the molecule number that contains
! atom number <i1>. If the atom is not part
! of any molecule zero is returned. The
! result is stored as res[1] as well.
! res[2] receives the position of the atom
! within the molecule.

rang(h1,h2,h3,k1,k2,k3[,l1,l2,l3])
! Returns the angle between vectors
! k-h and k-l at reciprocal site k.
! If l is omitted, the angle between the
! reciprocal vectors h and k is returned.

scalpro(u1,u2,u3,v1,v2,v3 [{,{"dd"|"rr"|"dr"|"rd"}}])
! Returns the scalar product between
! the two vectors u and v.
! Both vectors may be given in direct
! or real space coordinates, flagged
! in parameter no 7. "d" means real space,
! "r" reciprocal space.

gran(val,typ)
! Returns Gaussian distributed pseudo
! random number with mean zero and a width
! given by parameter <val>. If <typ> is
! "s" <val> is taken as sigma, if <typ>
! is "f", <val> is taken as FWHM.

gbox(r1,r2,r3)
! Returns pseudo random number with
! distribution given by a box centered
! at 0 with a width of <r2> and two half
! Gaussian distributions with individual
! sigmas of <r1> and <r3> to the left and
! right, respectively.

```

The arguments to any of these functions are any arithmetic expression.

## A.20 import

```

import "shelx",<file{"ins"}>
import "rmcprofile",<file{"cssr"}>

```

This command converts other crystal structure formats into a DISCUS unit cell.

**shelx****"shelx",<file{"ins"}>**

If the first parameter is "shelx" the input file is assumed to be a SHELX "\*.ins" or "\*.res" file. The extension ".ins" may be omitted. DISCUS will read the input file and create a file <file>".cell" in the DISCUS cell format. The space group is always set to P1, since SHELX defines the individual symmetry operations. The lattice is written into the DISCUS file as a generator. The SHELX symmetry operations are written as individual generator operations into the DISCUS file. The atom names are taken from the SHELX "SFAC" instruction and used for the individual atoms. Atom coordinates are properly interpreted including the use of SHELX free variables. LIMITATIONS The "DISP" instruction is ignored. Explicit form factors on the "SCAT" instruction are ignored ALL instructions following the "FVAR" instruction up to the "HKLF" instructions are interpreted as atoms, unless they are valid SHELX commands. The site occupation factors are ignored. The anisotropic atomic displacement parameters are crudely replaced by a very approximate isotropic U. Currently DISCUS does not recognize the internal symmetry of a molecule.

**rmcprofile**

DISCUS will read a RMCProfile \*.cssr file and write a corresponding structure file.

**A.21 index2cell****index2cell <atomindex> or****index2cell <x>,<y>,<z>,<site>**

This command allows to switch between a given index of the storage array <atomindex> and the corresponding unitcell <x>,<y>,<z> and site <site>. The parameter not given is calculated ...

index2cell stores the results in the "res[]" variable: res[1] : <atomindex> res[2] : <x> res[3] : <y> res[4] : <z> res[5] : <site>

The command is identical to the ==> 'trans' command in the chemistry menu.

**A.22 ins****ins <name>,<x>,<y>,<z> [,<t>]**

Inserts an atom of type <name> at the position <x,y,z> in crystal space. Optionally the temperature coefficient may be given.

**A.23 insert****insert {"domain"|"object"|"molecule"}**

Opens the menu to insert objects or molecules.

Objects are special molecules, that represent extended objects. DISCUS treats these objects as uniformly and continuously filled matter. Depending on the shape of the object, specialised Fourier transformations are calculated. Objects consist of four pseudo atoms with special names, that characterise the location, shape and orientation of the object.

Domain are special molecules as well, which are used to represent a guest structure within a host.

Further help topics are:

## commands

```
adp          ! Defines the atomic displacement parameter of an object
character    ! Defines the object/domain character = cube, cylinder, sphere
cent         ! Defines the center of atoms within a domain
density      ! Defines the scattering density of the object
file         ! Defines the file name with the actual domain content
fuzzy        ! Defines the distance between domain atoms and host atoms
origin       ! Places the origin of the object/domain in real space coordinates
run          ! Does the actual insertion
show         ! Show the current parameters
type         ! Defines the object type number
xaxis        ! Gives a deformation/scaling of the X-axis
yaxis        ! Gives a deformation/scaling of the Y-axis
zaxis        ! Gives a deformation/scaling of the Z-axis
xdim         ! Gives a deformation/scaling of the domain shape
ydim         ! Gives a deformation/scaling of the domain shape
zdim         ! Gives a deformation/scaling of the domain shape
```

## adp

**adp** <value>

## character

**character** {"cube"|"cylinder"|"sphere"}[,<atom name>]

Defines the character of the object, which may be one of {"cube"|"cylinder"|"sphere"}. Depending on the character, different formulae are used to calculate the Fourier transformation.

A "cube" object is defined by the six {100} faces and thus extends along each axis from -1. to +1. Its volume is thus eight times the volume of the unit cell. For a non-cubic unit cell, the faces are inclined accordingly.

A "cylinder" object has its cylinder axis along the [001] axis and extends from -1 to +1. The cross section in the x,y plane has a radius of 1. Thus its volume is  $2\pi$  times the unit cell volume. For non cubic unit cells, the cylinder will be ellipsoidal and possibly oblique. The top and bottom limiting ellipsoids are parallel to the two {001} faces.

A "sphere" object has a radius of 1. and a volume of  $\frac{4\pi}{3}$  times the unit cell. For non cubic unit cells, a general triaxial ellipsoid results.

The second, optional, parameter sets the atom type for the object. DISCUS will now assume that the object consists of atoms of this type, or at least that most of the atoms correspond to this type.

**cent**

**density**

**density** <value>

Defines the scattering density of the object type. The scattering factor calculated for objects of this type is multiplied by this density. This allows to simulated objects of identical volume yet different scattering density like bubbles of a denser material versus bubbles of empty space.

**file**

**fuzzy**

**origin**

**origin** <x>,<y>,<z>

The center of the object is placed at the vector <x>,<y>,<z>. DISCUS describes this position by the coordinates of the first of the atoms. It's name represents the 'character' of the object

**run**

Performs the actual insertion.

**show**

Shows the current settings

**type**

**type** {<number>|"new"}

Each object/molecule is assigned a type number and you may use this to perform the same operation on objects/molecules of identical type (like ==> symetry, shear, plot).

If <number> is given as a positive integer, the object/molecule will receive this number, which must correspond to an existing type.

If "new" is given, the object/molecule will receive the next higher unused type number.

**xaxis**

**xaxis** <x>,<y>,<z>

Defines in crystal space the scaling /and/or deformation of the x-axis. A set of values like 2,0,0 would mean a doubling of the objects x-axis.

DISCUS does not perform a check, whether the values that you entered correspond to any special deformation tensor. The ==> 'shear' menu provides elegant ways to deform an object.

**yaxis****yaxis** <x>,<y>,<z>

Defines in crystal space the scaling /and/or deformation of the y-axis. A set of values like 2,0,0 would mean a doubling of the objects x-axis.

DISCUS does not perform a check, whether the values that you entered correspond to any special deformation tensor. The ==> 'shear' menu provides elegant ways to deform an object.

**zaxis****zaxis** <x>,<y>,<z>

Defines in crystal space the scaling /and/or deformation of the z-axis. A set of values like 2,0,0 would mean a doubling of the objects x-axis.

DISCUS does not perform a check, whether the values that you entered correspond to any special deformation tensor. The ==> 'shear' menu provides elegant ways to deform an object.

**A.24 internal****Internal storage of the structure**

When reading a new cell, the program stores the content of each unit cell consecutively. The program always generates a block of  $n_x * n_y * n_z$  unit cells. The innermost -fastest- loop is  $n_x$ , the slowest  $n_z$ .

**A.25 inverse****inverse**

branches to the inverse Fourier transform level of DISCUS The inverse Fourier transform is calculated from two input files. The two files must contain the intensity, real or imaginary part amplitude or phase in one of the following combinations:

file a	file b
intensity	phase
amplitude	phase
real	imaginary

The user must specify the Fourier plane that had been calculated and the real space plane to be calculated.

Further help topics are:

**commands**

Valid commands at this level are:



```

@      ! execute a macro file (see main help level)
=      ! assigns the value to a variable (see main help level)
abs    ! defines the coordinate of the abscissa
continue! Continue a stopped macro (see main help level)
echo   ! echo a string (see main help level)
eval   ! Evaluates an expression (see main help level)
exit   ! terminates Fourier level
file   ! defines the input file name(s)
form   ! defines the input file format
help   ! help to Fourier commands (see main help level)
layer  ! defines the layer in reciprocal space to be calculated
ll     ! defines the lower left corner in reciprocal space
lr     ! defines the lower right corner in reciprocal space
na     ! defines the number of points along the abscissa
no     ! defines the number of points along the ordinate
ord    ! defines the coordinate of the ordinate
rhoabs ! defines the coordinate of the real space abscissa
rhoayer! defines the layer in real space to be calculated
rhoall ! defines the lower left corner in real space
rhoalr ! defines the lower right corner in real space
rhona  ! defines the number of points along the real space abscissa
rhono  ! defines the number of points along the real space ordinate
rhoord ! defines the coordinate of the real space ordinate
rhoal  ! defines the upper left corner in real space
run    ! starts the calculation
scale  ! defines an overall scale factor
set    ! sets various parameters
show   ! shows the current settings for the inverse Fourier transform
stop   ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
ul     ! defines the upper left corner in reciprocal space
wait   ! Waits for user input (see main help level)

```

## abs

**abs <switch>**

Defines which the component of the vector parallel to the abscissa is written to the output file.

## file

**file {"a"|"b"},<filename>**

Defines the filenames needed for the inverse Fourier.

The inverse Fourier transform is calculated from two input files. The two files must contain the intensity, real or imaginary part amplitude or phase in one of the following combinations:

```

file a          file b
-----
intensity
amplitude
real           imaginary

```

If the input format is 'shelxl' then one input file is sufficient. DISCUS assumes the input file to be a SHELXL List type 5 containing <h> <k> <l> <Fobs> <Fcalc> <phase angle in degree> The file is read as free format.

**format****form** {"gnu" | "stan" | "shelxl"}

Defines the input format that will be used. Valid parameters are:

**"gnu"**

Sets the type of the output file to gnuplot. The intensities are written in blocks of intensities along the abscissa, with an empty line separating each block. Each line of each block contains: index along abscissa, index along ordinate, intensity, third index.

**"stan"**

Sets the output to standard file format, which is used by KUPL. The output file contains two initial lines and then a block for each data line separated by an empty line. The first line contains the number of data points along the abscissa and the ordinate, the second line the minimum and maximum value of the indices in reciprocal space along the abscissa and ordinate respectively.

**"shelxl" [, {"fobs" | "fcalc"}]**

Sets the input file format to SHELXL. DISCUS assumes the input file to be a SHLEXL List type 5 containing <h> <k> l> <Fobs> <Fcalc> <phase angle in degree> The file is read as free format. All reflections are expanded by the appropriate symmetry operations, including proper phase transformation.

If the optional parameter is "fobs", the observed structure factor is used for the calculation, if the parameter is "fcalc", the calculated structure factor is used.

**layer****layer** <e11,e12,e13, e21,e22,e23, e31,e32,e33, inc1,inc2>

sets the lower left, lower right and upper left corner of the Fourier plane to be calculated and sets the number of data points along the horizontal and the vertical

**ll****ll** <e11> , <e12> , <e13>

Sets the lower left corner of the Fourier plane to be calculated.

**lr****lr** <e21> , <e22> , <e23>

Sets the lower right corner of the Fourier plane to be calculated.

**na****na** <inc1>

Sets the number of data points that will be calculated along the abscissa.

**no**

**no** <inc2>

Sets the number of data points that will be calculated along the ordinate.

**ord**

**ord** <switch>

Defines which the component of the vector parallel to the ordinate is written to the output file.

**rhoabs**

**rhoabs** <switch>

Defines which the component of the vector parallel to the real space abscissa is written to the output file.

**rholayer**

**rholayer** <e11,e12,e13, e21,e22,e23, e31,e32,e33, inc1,inc2>

sets the lower left, lower right and upper left corner of the real space plane to be calculated and sets the number of data points along the horizontal and the vertical

**rholl**

**rholl** <e11> , <e12> , <e13>

Sets the lower left corner of the real space plane to be calculated.

**rhplr**

**rhplr** <e21> , <e22> , <e23>

Sets the lower right corner of the real space plane to be calculated.

**rhona**

**rhona** <inc1>

Sets the number of data points that will be calculated along the real space abscissa.

**rhono**

**rhono** <inc2>

Sets the number of data points that will be calculated along the real space ordinate.

**rhoord****rhoord** <switch>

Defines which the component of the vector parallel to the real space ordinate is written to the output file.

**rhoul****rhoul** <e31> , <e32> , <e33>

Sets the upper left corner of the real space plane to be calculated.

**run****run**

Starts the inverse Fourier calculation.

**scale****scale** <scale-factor>

Defines an overall scale factor. Together with the automatic scaling by 1/vol(unit cell), this scale factor serves to put the electron density on an absolute scale of electrons per cubic Angstrom. The scale factor should effectively multiply the observed F(000) such that it is equal to F(000) calculated for one unit cell of the average structure.

Every time you enter "inverse" the scale factor is calculated to be number of atoms in crystal / number of atoms in one unit cell. If you use experimental data, you MUST specify the scale factor every time you enter the "inverse" submenu.

**set****set** "accu",{ "init"|"add" }**set** "excl",{ none|<value> }**set** "rsym",{ appl|"igno" }

Set defines various parameters.

**set** "accu",{ "init"|"add" }

Sets the mode for the accumulation of inverse Fourier transforms.

**"init"**

The electron density array is initialized, i.e. set to zero.

**"add"**

The next inverse Fourier transform is added to the previous value. This way several inverse Fourier transforms will contribute to a single layer in direct space, which is necessary to create a inverse Fourier section rather than a projection onto the layer. See the manual for further details.

If you use DISCUS to calculate the input data for the inverse Fourier, you MUST separate the loop that calculates the Fourier from the loop that calculates the inverse Fourier!!! Both parts of the program use the same variable to store the results and by mixing the loops, the result will be unpredictable nonsense !

**"excl",{"none"<value>}**

Signals whether certain data points should be ignored.

**"none"**

All data points are treated as normal data points, regardless of their value. This is the default at program start.

**<value>**

Data points that are equal to <value> are excluded from the inverse Fourier transform. You can use this option to mask out regions of the input data by setting their values to <value>. In the companion program KUPLOT, data points of value equal to -9999. are treated as excluded regions.

**set "rsym",{"appl"|"igno"}**

Determines whether the space group symmetry is applied to all reflections. If the second parameter is "appl", all symmetry elements are applied to all reflections and the phases are transformed accordingly. This allows you to read a set of unique reflections and calculate the inverse Fourier.

**show**

**show**

shows the current settings for the Fourier transform file

**type**

**type {"a"|"b"},  
{"intensity"|"amplitude"|"real"|"imaginary"|"phase"|"fobs"|"fcalc"}**

Defines the content of file "a" or "b".

The inverse Fourier transform is calculated from two input files. The two files must contain the intensity, real or imaginary part amplitude or phase in one of the following combinations:

file a	file b
intensity	phase
amplitude	phase
real	imaginary

If the input format is "shelxl", DISCUS reads just one file. You can calculate the inverse Fourier from Fobs or Fcalc data by choosing the appropriate file type from one of the following choices:

```

file a          file b
-----
amplitude
fobs
fcalc

```

**ul**

**ul** <e31> , <e32> , <e33>

Sets the upper left corner of the Fourier plane to be calculated.

## A.26 kick

**kick** <name>, <x>,<y>,<z>,<t>,<na>,<ne>,<delx> [,<dely> [,<delz>]]

**kick** <name>, <x>,<y>,<z>,<t>,<na>,<ne>,-<bondlength>

Inserts an atom of type <name> at the position <x,y,z> in crystal space. The temperature coefficient must be given. 'kick' checks the position <x>,<y>,<z> with respect to all atoms numbered <na> to <ne>. All atoms that are within the block <x> +- <delx>; <y> +- <dely>; <z> +- <delz> are removed from the structure.

If <dely> and/or <delz> are not given, then they default to the value of the last <del.> given on this 'kick' command.

Note that the <del.> are in fractional coordinates, NOT in Angstrom.

In the alternative form, the eighth's parameter is interpreted as a bond distance. To distinguish the two forms, the bondlength must be less than zero. The new atom is inserted and all atoms within <bondlength> Angstrom from the position <x>,<y>,<z> are removed.

The parameters <na> and <ne> must be in the range 1 to current number of atoms in the crystal i.e. the value of variable "n[1]", and <na> must be less or equal to <ne>. The one exception is <na> = n[1]+1 and <ne> = n[1]. This allows you to start building e.g. a new layer within which you would like to optionally insert atoms without affecting the previous layer. Example i[0] = n[1]+1 do i[1]=1,10 insert al,ran(0),ran(0),0.5, 1.0, i[0],n[1], -2.4 enddo This loop inserts up to 10 Al atoms in the x,y range 0 to 1 at z=0.5 with a minimum distance of 2.4 Angstrom. Since i[0], the value for <na> is initialla at n[1]+1, all previously inserted atoms are not affected. Since i[0] remains fixed within the loop, each new al atom affects all al atoms previously inserted within this loop.

## A.27 mc

**mc**

NOTICE: The "mc" level has been depreciated and is no longer supported. Please move to the "mmc" level, which has the same functionality plus lots of extra possibilities.

The "mc" level has been removed in this version of the program.

Enter "Monte Carlo" section of DISCUS. This sublevel allows to create disordered structures with given correlations and/or displacements using MC simulations. See help entry 'commands' for a complete list of commands available in this sublevel.

NOTE: The MC and CHEM sublevel share some variables which define neighbours etc. and settings made in this sublevel might be altered when using the other sublevel. It is always save to repeat settings when entering this sublevel if both levels are used.

## commands

Valid commands at this sub level are:

```
@      ! Executes a macro (see main help)
=      ! Algebra (see main help)
continue! Continue a stopped macro (see main help level)
echo   ! Echos a string, just for interactive check
eval   ! Evaluates an expression (see main help)
exit   ! Terminates the MC sub level, returns to the main DISCUS level.
help   ! Gives on-line help for 'mc' (see main help)
run    ! Starts MC simulation
save   ! Saves structure (see main help -> save)
set    ! Sets most MC parameters
show   ! Shows current settings
stop   ! Stops execution of a macro (see main help level)
system ! executes operating system command (see main help)
wait   ! Waits for user input (see main help)
```

## run

### run

This command starts the MC simulation for the given parameters and number of MC moves.

## set

### set <subcommand>

This command allows to define the wanted correlations and most other parameters for the MC simulation. The help entry 'commands' contains a list of valid subcommands for the 'set' command. Further help can be found for the following subcommands:

## commands

```
angle   : Definition of interaction angles
atom    : Setting of atom types for MC simulation
const   : Setting of interaction constants
cyc     : Setting of maximum number of MC moves
energy  : Setting of MC mode (energy calculation)
feed    : Setting of display/feedback interval
mode    : Setting of MC mode (crystal modification)
mole    : Setting molecule types for MC simulation
move    : Setting size of generated SHIFT moves
neig    : Setting of the neighbour definitions
target  : Setting of correlations values to be achieved
temp    : Setting of MC simulation temperature
vec     : Definition of interaction vectors
```

**angle**

**set "ang",<iv>,<is1>,<is2>,<dx1>,<dy1>,<dz1>,<is3>,<dx2>,<dy2>,<dz2>**

This command is used to define correlation angles that might be used to calculate correlations within the crystal (-> 'set corr'). The complete description is given in the CHEM section of the online help (-> set vec).

**atom**

**set "atom",{ <name> | <number> }, { <name> | <number> }  
[, { <name> | <number> }]**

This command allows the user to select the atoms types used for the MC simulation. They can either be specified by <name> (e.g. Zr) or <number>. Not that different MC energy definitions have different restrictions on the selection of the atoms. The angular distortions need three angles, while the other ones need just two.

**const**

**set "const",<ic>,<const> [,<fac>]**

The introduction of certain correlations depend on a given set of near neighbour interactions. These values are initially unknown and are adjusted using a feedback method (-> set feed). This command allows the user to set initial interaction parameters <const> for the corresponding interaction/correlation <ic>. The optional last parameter <fac> is a factor used in the feedback process and controls how 'fast' the interactions are changed in each feedback cycle.

**cyc**

**set "cyc",<number>**

This command sets the number of MC moves (i.e. visited sites) to be executed by the 'run' command. The number <number> actually defines the number of 'tried' MC moves. Note that we refer to a MC cycle as the number of moves needed to visit every crystal site once on average.

**energy**

**set "energy",{ "angle" | "cocc" | "cdis" | "disp" }**

This command selects the type of MC energy to be used for the simulation. The MC sublevel of DISCUS currently allows to introduce occupational correlations ("cocc"), displacement correlations ("cdis"), distortions ("disp") and angles ("angle"). The energy for the correlation modes is based on an Ising model whereas the distortions are expressed using Hookes law. The angular distortions are expressed as the square of the deviation from the target value.

**feed**

**set "feed",<number>**



The interaction values for the given correlation structure are determined by a feedback process. This command sets the number of moves between each feedback run. This value is also used for the interval of screen outputs. Note that the energy mode "disp" does not use the feedback mechanism and the value <number> given here only determines the output interval.

### mode

```
set "mode",{ "shi[ft]" | "swc[hem]" | "swd[isp]" | "ext[ernal]" },
[ { "a[l]" | "l[ocal]" | "sl[ocal]" | "si[te]" } ]
```

This command determines the type of MC move used for modifying the crystal. The settings are explained in the RMC section under -> set mode.

### neig

```
set "neig","add"
set "neig","ang",<iv1> [,<iv2>,..]
set "neig","dis",<u>,<v>,<w>,<fsig> [, <wsig>] [, {"sym" | "nosym"}]
set "neig","dir",<x1>,<y1>,<z1> [,<x2>,<y2>,<z2>]
set "neig","rese"
set "neig","vec",<iv1> [,<iv2>,..]
```

This command is used to define the neighbours used to determine the correlations. The subcommand 'dis' uses the distance as a criterion whereas 'vec' uses defined interaction vectors (-> set vec). The subcommand 'ang' uses an angle at a site, defined by (-> set ang). The subcommands 'add' and 'rese' allow the storage and reset of the list of definitions. The details of these commands are given in the online help for the same command 'set neig' in the CHEM sublevel.

### mole

```
set "mole",<typ1>,<typ2>
```

This command allows the user to select the molecule types used for the MC simulation. Note that different MC energy definitions have different restrictions on the selection of the molecule types.

### move

```
set "move",{ "all" | <name> | <number> },<sx>,<sy>,<sz>
```

Sets an user defined sigma for the generates moves. The created shifts are Gaussian distributed. The default is a sigma of 0.2 unit cell. Note that the values of <sx>, <sy> and <sz> are given in unit cell units to speed up the MC runs. In case molecules are selected <number> specifies the molecule type, otherwise atoms are addressed by their <number> or <name>.

**range**

**target**

**set "target",<ic>,<value>**

**set "target",<ic>,{<number> | <name> | "all"},  
{<number> | <name> | "all"}, <dist>**

The first command sets the desired correlation value <value> for modes "cocc" and "cdis" for the neighbour definition <ic>.

The second version of the command sets the desired distance between the given atom types to <dist> Å. This setting is used for mode "disp". If <disp> is set to zero the corresponding atom pair is not used in the MC simulation.

**temp**

**set "temp",<kt>**

This command sets the temperature <kt> used for the MC simulation. In order to be able to use the feedback procedure to determine the interaction energies the value of <kt> should be set to 1.0 (default). As a result the interaction energies will be in units of kT. If no feedback is used, <kT> can be set to the desired simulation temperature.

**vec**

**set "vec",<iv>,<is1>,<is2>,<dx>,<dy>,<dz>**

This command is used to define correlation vectors that might be used to calculate correlations within the crystal (-> 'set corr'). The complete description is given in the CHEM section of the online help (-> set vec).

**show**

**show**

This command lists all current settings for the MC simulation part of DISCUS on the screen.

## A.28 mmc

**mmc**

Enter "Multi Energy Monte Carlo" section of DISCUS. This sublevel allows to create disordered structures with given correlations and/or displacements using Monte-Carlo simulations. Several different "energies" can be minimized simultaneously.

See help entry 'commands' for a complete list of commands available in this sublevel.

NOTE: The MMC, MC and CHEM sublevel share some variables which define neighbours etc. and settings made in this sublevel might be altered when using the other sublevel. It is always save to repeat settings when entering this sublevel if both levels are used.

NOTE: Many of the commands of the MMC mode are at first sight similar to the corresponding MC commands.

The mmc menu modifies the crystal according to the moves, ==> 'set move' that you define. Only those atoms will take part of the 'moves' and be included in the energy calculations, whose properties fulfill the global property selection rules ==> 'property'.

## commands

Valid commands at this sub level are:

```
@      ! Executes a macro (see main help)
=      ! Algebra (see main help)
continue! Continue a stopped macro (see main help level)
echo   ! Echos a string, just for interactive check
eval   ! Evaluates an expression (see main help)
exit   ! Terminates the MMC sub level, returns to the main DISCUS level.
help   ! Gives on-line help for 'mmc' (see main help)
rese   ! Reset the MMC to startup values
run    ! Starts MMC simulation
save   ! Saves structure (see main help -> save)
set    ! Sets most MMC parameters
show   ! Shows current settings
stop   ! Stops execution of a macro (see main help level)
system ! executes operating system command (see main help)
wait   ! Waits for user input (see main help)
```

## rese

### rese

Resets the value of all internal variables, counters, etc to the startup default values.

## run

### run

This command starts the MMC simulation for the given parameters and number of MMC moves.

## set

### set <subcommand>

This command allows to define the wanted correlations and most other parameters for the MMC simulation. The help entry 'commands' contains a list of valid subcommands for the 'set' command. Further help can be found for the following subcommands:

## commands

```
allowed : Define which atom types are part of a neighborhood
angle   : Definition of interaction angles
con     : Definitions of interaction connectivities
cyc     : Setting of maximum number of MMC moves
envir   : Definitions of interaction environments
feed    : Setting of display/feedback interval
fixed   : Setting of atom ranges that are fixed
```

```

limit      : Setting of a limited atom range
mode       : Setting of MMC mode (crystal modification)
move       : Setting size of generated SHIFT moves
neig       : Setting of the neighbour definitions
range      : Setting of the neighbouring ranges
target     : Setting of correlations values to be achieved
temp       : Setting of MMC simulation temperature
vec        : Definition of interaction vectors

```

## allowed

**set "allowed", <is1> [, <is2>]...**

Usually mmc will automatically determine, which atom types play a role in the Monte-Carlo process. To do so mmc analyzes the ==> 'set target' command to find out which atoms are involved.

In some cases, you might have defined a target simply as "set target, 1,repulsive,void,void, ..." Here the purpose is to define a repulsive interaction between voids, while all other atoms do not play a role. MMC needs to know, however, whether it may shift/switch chemistry of voids with other atoms through the "set mode" definitions.

To allow atoms to take part in the "set mode" modifications of the crystal, although they do not influence the energy of the crystal, you need to list these atoms on this "set allowed" command.

## angle

**set "ang", <iv>, <is1>, <is2>, <dx1>, <dy1>, <dz1>, <is3>, <dx2>, <dy2>, <dz2>**

This command is used to define correlation angles that might be used to calculate correlations within the crystal (-> 'set target'). The complete description is given in the CHEM section of the online help (-> set ang).

## con

**set "con", <iv>, {<is1> | <at\_name>}, {<def.no> | <def.name>}**

This commands is used to define correlations between atoms through their connectivity list. See main menu 'connectivity' for further details. The second and third parameter allow alternative input. <iv> is simply the number of the connectivity definition within mmc. One or several connectivity definitions can be grouped to a neighborhood.

<is1> is the atom type of the central atom around which neighbors shall be searched via the connectivity list. <at\_name> is the name of the central atom type around which neighbors hall be searched via the connectivity list.

<def.no.> is the number of the definition that has been added/set within the connectivity menu. <def.name> is the name of the definition that has been added/set within the connectivity menu.

## cyc

**set "cyc", <number>**

This command sets the number of MMC moves (i.e. visited sites) to be executed by the 'run' command. The number <number> actually defines the number of 'tried' MMC moves. Note that we refer to a MMC cycle as the number of moves needed to visit every crystal site once on average.

### disallowed

**set "disallowed", <is1> [,<is2>...]**

Usually mmc will automatically determine, which atom types play a role in the Monte-Carlo process. To do so mmc analyzes the ==> 'set target' command to find out which atoms are involved.

Sometimes, you might want to fix an atom to a place within the crystal structure. At the same time, though you might want the correlations between this atom and other atoms to be taken into account. This is needed, for example if you want to sort atoms of type A into the center of a sphere, while keeping atoms of type B on the outside. In this case a dummy atom of type C should be fixed at the center, and negative chemical short range order be realized between this C atom and atoms of type A.

To allow an atom type to participate in the moves defined by ==> set move, use the set allowed command.

As the ==> 'set target' command allows all atoms that it finds to participate in a move, the 'set disallowed' command must be placed after the last 'set target' instruction.

### environment

**set "env",**

### feed

**set "feed",<number>**

The interaction values for the given correlation structure are determined by a feedback process. This command sets the number of moves between each feedback run. This value is also used for the interval of screen outputs. Note that the energy mode "disp" does not use the feedback mechanism and the value <number> given here only determines the output interval.

### limited

**set "limited", "OFF"**

**set "limited", "atom", <nmin>, <nmax>**

Allows to limit the atom selection the interval [<nmin>:<nmax>]. Atoms outside this interval are ignored and thus do not take part in the Monte Carlo process.

The "set limit, OFF" command switched the limitation off.

### mode

**set "mode",<prob>,{ "shi[ft]" | "swc[hem]" | "swd[isp]" | "ext[ernal]" },**

```
[{ "a[l]" | "l[ocal]" | "sl[ocal]" | "si[te]" }
, [<atom_list>] ]
```

This command determines the type of MMC move used for modifying the crystal. You can simultaneously use any of the moves. The user defined values of the probabilities <prob> are internally scaled to add up to one and give the probability for each type of move. The settings are explained in the RMC section under -> set mode. If the atom list is given, the fourth parameter that specifies the site must also be present. The atom list allows you to "move" or "exchange" atoms, even if they are not involved in any of the target energies.

### move

```
set "move",{ "all" | <name> | <number> },<sx>,<sy>,<sz>
```

Sets an user defined sigma for the generated moves. The created shifts are Gaussian distributed. The default is a sigma of 0.2 unit cell. Note that the values of <sx>, <sy> and <sz> are given in unit cell units to speed up the RMC runs. In case molecules are selected <number> specifies the molecule type, otherwise atoms are addressed by their <number> or <name>.

### neig

```
set "neig","add"
set "neig","ang",<iv1>,{ [<iv2>,..] | <iv2>,"range" }
set "neig","env",<iv1>,{ [<iv2>,..] | <iv2>,"range" }
set "neig","rese"
set "neig","vec",<iv1>,{ [<iv2>,..] | <iv2>,"range" }
```

This command is used to define the neighbours used to determine the correlations. The subcommand 'env' uses defined interaction environments (-> set env). The subcommand 'vec' uses defined interaction vectors (-> set vec). The subcommand 'ang' uses an angle at a site, defined by (-> set ang). The subcommands 'add' and 'rese' allow the storage and reset of the list of definitions. The details of these commands are given in the online help for the same command 'set neig' in the CHEM sublevel.

If the last parameter is "range", and exactly two arguments <iv1>,<iv2> are given, then the range of correlations from <iv1> to <iv2> is used.

Examples set neig, vec, 1,2,8,7 set neig, vec, 1,2,3,4,5,6 set neig, vec, 1,6,range

### move

```
set "move",{ "all" | <name> | <number> },<sx>,<sy>,<sz>
```

Sets an user defined sigma for the generated moves. The created shifts are Gaussian distributed. The default is a sigma of 0.2 unit cell. Note that the values of <sx>, <sy> and <sz> are given in unit cell units to speed up the MMC runs. In case molecules are selected <number> specifies the molecule type, otherwise atoms are addressed by their <number> or <name>.

range

set "range"

target

```
set "target",<ic>,"corr", {<number> | <name> | "all"},
{<number> | <name> | "all"},
<correlation>, <energy>, {"CORR" | "ENER"}
set "target",<ic>,"corr", ({<number> | <name> | "all"}[, {<number> | <name> | "all"}...]),
({<number> | <name> | "all"}[, {<number> | <name> | "all"}...]),
<correlation>, <energy>, {"CORR" | "ENER"}
set "target",<ic>,"cd", {<number> | <name> | "all"},
{<number> | <name> | "all"},
<correlation>, <energy>, {"CORR" | "ENER"}
set "target",<ic>,"angle", {<number> | <name> | "all"},
{<number> | <name> | "all"},
{<number> | <name> | "all"},
<angle>, <energy>
set "target",<ic>,"spring", {<number> | <name> | "all"},
{<number> | <name> | "all"},
<distance>, <energy>
set "target",<ic>,"lennard", {<number> | <name> | "all"},
{<number> | <name> | "all"},
<distance>, <energy> [, <pr>, <pa>]
set "target",<ic>,"bucking", {<number> | <name> | "all"},
{<number> | <name> | "all"},
<a>, <rho>, <b>
set "target",<ic>,"repulsive",{<number> | <name> | "all"},
{<number> | <name> | "all"},
<energy>, <scale> [, <rmin> [, <pa>] ]
```

The target command sets the desired correlation values for each of the defined interaction neighbourhoods <ic> (-> set neigh).

Depending on the neighbourhood type you must specify exactly one or several target energy types for each <ic>. The neighbourhoods "vector", <cor>, and "angle" require one energy type, while the neighbourhood "environment" may take several different energies. For each interaction <ic> you may specify several targets for different atom combinations.

```
set "target",<ic>,"corr", {<number> | <name> | "all"}, {<number> | <name> | "all"}, <correlation>,
<energy>, {"CORR" | "ENER"} set "target",<ic>,"corr", ({<number> | <name> | "all"}[, {<number>
| <name> | "all"}...]), ({<number> | <name> | "all"}[, {<number> | <name> | "all"}...]),
```

A chemical short range order correlation is defined between the two atom types. The last keyword defines how the target shall be achieved. With "CORR", you want to achieve the target <correlation>. DISCUS uses the feedback loop to adjust the value of the Ising Energy in order to achieve the desired correlation. With "ENER", you fix the value of the Ising energy to the value of <energy>. For negative <energy> values like atoms will tend to be neighbors, while

a positive <energy> favors opposite atoms next to each other. If you use several targets at the same time, it is necessary to use the "Ener" keyword. Otherwise, the feedback mechanism will increase the absolute value of the Ising energy at each feedback until the target <correlation> is achieved. This target will therefor overrule the other targets that all have fixed energy values. If you specify a positive correlation, like tom types will tend to cluster together, while for a negative correlation you will get mostly opposite atom types as first neighbors.

The second command form allows you to define correlations between groups of atom types. Place each of the two sets of atom types in round brackets. In this case a positive correlation will lead to clusters that consist of any atom of the first group to be neighbor to another atom of the first group and likewise at a different place in the crystal for the second group. A negative correlation will give you a crystal in which an atom of the first group has an atom of the second group as neighbor.

```
set "target",<ic>,"lennard", {<number> | <name> | "all"}, {<number> | <name> | "all"},
<distance>, <energy> [, <pr>, <pa>]
```

A Lennard Jones type potential energy is calculated for the two atom types. The energy is calculated as:  $E = A / \langle r \rangle^{12} - B / \langle r \rangle^6$ . Here <r> is the interatomic distance and A and B are calculated to give the minimum of the Lennard-Jones potential at distance <distance> with a potential depth of -1.\*|<energy>|

```
set "target",<ic>,"repulsive",{<number> | <name> | "all"}, {<number> | <name> | "all"},
[<energy>[,<scale> [, <rmin> [, <pr>] ]]]
```

A purely repulsive energy is calculated between the two atom types. The energy is calculated as:  $-1.*|<energy>| + ((\langle distance \rangle - \langle rmin \rangle) / \langle scale \rangle)^{12}$ . Thus <energy> is the energy between the two atoms at infinite distance. As mmc compares the energy differences between two atom configurations, the value of this energy is not really relevant.

For interatomic distances shorter than <rmin> the program calculates:  $-1.*|<energy>| + ((0.000001) / \langle rmin \rangle)^{12}$

For a steep descent, choose a high value of the power <pr>. With a high value of <pr>, the potential levels off at shorter distances and energy differences become less relevant. Thus the atoms will be pushed apart more strongly at shorter distances, yet at longer distances the influence of this potential is less critical.

To push atoms far apart over the whole range of distances, choose a large value of <scale>.

<energy> defaults to zero <scale> defaults to 1.0 <rmin> defaults to zero <pr> defaults to 1.0

## temp

```
set "temp",<kt>
```

This command sets the temperature <kt> used for the MMC simulation. In order to be able to use the feedback procedure to determine the interaction energies the value of <kt> should be set to 1.0 (default). As a result the interaction energies will be in units of kT. If no feedback is used, <kT> can be set to the desired simulation temperature.

## vec

```
set "vec",<iv>,<is1>,<is2>,<dx>,<dy>,<dz>
```



This command is used to define correlation vectors that might be used to calculate correlations within the crystal (-> 'set neig'). The first parameter <iv> is the number of the vector to be defined. The variables <is1> and <is2> give the number of the crystal site at each end of the vector. The values of <dx>, <dy> and <dz> define the vector in unit cells. Correlation vectors within one unit cell have <dx>, <dy> and <dz> set to zero. If you want to define correlation vectors crossing the unit cell boundaries set <dx>, <dy> and <dz> accordingly.

## show

### show

This command lists all current settings for the MC simulation part of DISCUS on the screen.

## A.29 output

### output

Switches to the output level of discus. All Fourier output is written at this level. Several graphic formats are available.

Further help topics are:

### commands

```
@      ! Execute a macro file (see main help)
=      ! assigns the value to a variable (see main help)
continue! Continue a stopped macro (see main help level)
echo   ! Echo a string (see main help)
eval   ! Evaluates an expression for interactive check (see main help)
exit   ! Returns to the main discus level
form   ! Defines the output format to be used
help   ! Gives help for the output level (see main help)
input  ! reads an old data file
outfile ! sets name of output file
run    ! Starts writing of output
show   ! Shows the current settings
thre   ! Sets the thresholds used for bitmap output
valu   ! Sets the value that is written to file
stop   ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
wait   ! Waits for user input (see main help)
```

### format

```
form {"gnu" | "pgm" | "ppm" | "powder" [, {"dst" | "tth" | "stl" | "q"}]
| "post" | "shel" | "stan" | "hklf4" | "list5" }
```

Defines the output format that will be used. Valid parameters are:

**"gnu"**

Sets the type of the output file to gnuplot. The intensities are written in blocks of intensities along the abscissa, with an empty line separating each block. Each line of each block contains: index along abscissa, index along ordinate, intensity, third index.

**"list5"**

Sets the output type to a SHELXL list 5. Each line of the output contains "h k l F phase" in the format (3I4, 2F10.2,F2.2). The hkl are converted by an "int" command to integers. The user has to make sure that the increments used in the Fourier resulted in integer spacing. Independent of the output value set by the 'value' command, DISCUS always writes the amplitude and the phase angle in degrees. DISCUS calculates the value of F(000) and scales the data such that F(000) is less than  $10^8$  to fit into the format.

**"pgm"**

Sets the type of the output to portable grey map. A bitmap of the intensity is written in portable grey map style. All values below the minimum threshold are set to this value, all values above the maximum threshold are set to this maximum value.

**"ppm"**

Sets the type of the output to portable any-map. A bitmap of the intensity is written in portable any-map style. The color scheme is read from file "color.map". All values below the minimum threshold are set to this value, all values above the maximum threshold are set to this maximum value.

**"powder"[,{"dst"|"tth"|"stl"|"q"}]**

Sets the output format to powder. You must have calculated a powder diffraction pattern using the ==> 'powder' menu of DISCUS. DISCUS writes a two column output, with the intensity in the second column. The value in the first column is set by the optional second parameter and may be: "dst" :  $dstar = 1./d(hkl) = 2 * \sin(\theta) / \lambda$  "tth" :  $2-\theta$  "stl" :  $\sin(\theta) / \lambda$  "q" :  $Q = 2 * \pi * dstar = 4 * \pi * \sin(\theta) / \lambda$

For the two types of axes in the ==> 'powder' menu, "Q" and "TTH", you need to remember to specify limits for the axis. If you intend to write the output in the other format, you need to specify the limits on this axis as well.

**"post"**

Sets the type of the output to color postscript. A bitmap of the intensity is written in color postscript. The color scheme is read from file "color.map". All values below the minimum threshold are set to this value, all values above the maximum threshold are set to this maximum value.

**"shel"****"hklf4"**

Both commands sets the output format to SHELX type hklf4 data. The data are written in format (3I4,2F8.2). The hkl are converted by an "int" command to integers. The user has to make sure that the increments used in the Fourier resulted in integer spacing. Use the ==> 'value' "intensity" or "amplitude" for the output. DISCUS calculates the value of I(000) or F(000) respectively and scales the data such that I(000) is less than  $10^6$  to fit into the format. The sigma is just the square root of the final intensity after scaling.

**"stan"**

Sets the output to standard file format, which is used by KUPLOT. The output file contains two initial lines and then a block for each data line separated by an empty line. The first line contains the number of data points along the abscissa and the ordinate, the second line the minimum and maximum value of the indices in reciprocal space along the abscissa and

ordinate respectively.

## input

### input

Reads an old data file

## outfile

### outfile <filename>

Sets the name of the output file, to which the intensities are written.

## run

### run

Starts writing the output file. Without this command the output file is not written!

## show

### show

Shows the current settings

## threshold

### thresh { "high"|"low"|"sigma"|"zmin"|"zmax" } <value>

This command sets the threshold that is used when writing BITMAP output. All values less than the minimum threshold are set to zero, all values higher than the maximum threshold are set to the maximum threshold. The values in between are linearly scaled from zero to 255.

Depending on the first parameter, the second parameter is interpreted in five different ways:

```
"high" : Sets maximum threshold for BITMAP in percent of the maximum
         diffuse intensity.
"low"   : Sets minimum threshold for BITMAP in percent of the maximum
         diffuse intensity.
"sigma" : Sets threshold for BITMAP to average diffuse intensity
         +- <value> times standard deviation of diffuse intensity.
"zmax"  : Sets maximum threshold for BITMAP
"zmin"  : Sets minimum threshold for BITMAP
```

## value

### valu { "int" | "amp" | "pha" [, "random"] | "real" | "imag" }

### valu { "<int>" | "<amp>" | "<pha>" [, "random"] | "<real>" | "<imag>" }

Sets what value is written to the output file. The keyword given in < > corresponds to the value of the average structure factor <F> whereas the plain word stands for the resulting value of the Fourier transform which can be F or F-<F> depending on the selected modus. The command

'value int' will save the resulting intensity whereas 'value <int>' will give the intensity of the average structure. The allowed values are summed up in the following list:

```
"int" :   intensity I(hkl)
"amp" :   modulus of structure factor |F(hkl)|
"pha" :   phase angle in degrees
           If the optional second parameter is given as "random",
           the phases for integer HKL are written as calculated.
           All other phases are assigned a random number between
           -180 and +180 degrees.
"real":   real part of structure factor
"imag":   imaginary part of structure factor
```

The file name is not adapted automatically!

## A.30 patterson

### patterson

branches to the Patterson transform level of DISCUS The Patterson transform is calculated from two input files. The two files must contain the intensity, real or imaginary part, or amplitude in one of the following combinations:

```
file a           file b
-----
intensity
amplitude
real             imaginary
```

The user must specify the Fourier plane that had been calculated and the real space plane to be calculated.

Further help topics are:

### commands

Valid commands at this level are:

```
@      ! execute a macro file (see main help)
=      ! assigns the value to a variable (see main help)
abs     ! defines the coordinate of the abscissa
continue! Continue a stopped macro (see main help level)
echo    ! echo a string (see main help)
eval    ! Evaluates an expression for interactive check (see main help)
exit    ! terminates Fourier level
file    ! defines the input file name(s)
form    ! defines the input file format
help    ! help to Fourier commands (see main help)
layer   ! defines the layer in reciprocal space to be calculated
ll      ! defines the lower left corner in reciprocal space
lr      ! defines the lower right corner in reciprocal space
na      ! defines the number of points along the abscissa
no      ! defines the number of points along the ordinate
ord     ! defines the coordinate of the ordinate
rhoabs  ! defines the coordinate of the real space abscissa
rhoayer! defines the layer in real space to be calculated
rhoall  ! defines the lower left corner in real space
rhoalr  ! defines the lower right corner in real space
```

```

rhona    ! defines the number of points along the real space abscissa
rhono    ! defines the number of points along the real space ordinate
rhoord   ! defines the coordinate of the real space ordinate
rhoul    ! defines the upper left corner in real space
run       ! starts the calculation
scale    ! defines an overall scale factor
set       ! sets various parameters
show     ! shows the current settings for the inverse Fourier transform
stat     ! Shows statistics on the normalized structure factor
stop     ! Stops execution of a macro (see main help level)
system   ! Executes operating system command (see main help)
type     ! defines the file type (intensity, amplitude ...)
ul       ! defines the upper left corner in reciprocal space
wilson   ! Calculates Wilson statistics
wait     ! Waits for user input (see main help)

```

## abs

**abs** <switch>

Defines which the component of the vector parallel to the abscissa is written to the output file.

## file

**file** {"a"|"b"},<filename>

Defines the filenames needed for the patterson.

The Patterson transform is calculated from one or two input files. The two files must contain the intensity, real or imaginary part, or amplitude in one of the following combinations:

```

file a           file b
-----
intensity
amplitude
real             imaginary

```

If the input format is 'shelxl' then one input file is sufficient. DISCUS assumes the input file to be a SHLEXL List type 5 containing <h> <k> l> <Fobs> <Fcalc> <phase angle in degree> The file is read as free format.

## format

**form** {"gnu" | "stan" | "shelxl" | "hk1f4" }

Defines the input format that will be used. Valid parameters are:

**"gnu"**

Sets the type of the output file to gnuplot. The intensities are written in blocks of intensities along the abscissa, with an empty line separating each block. Each line of each block contains: index along abscissa, index along ordinate, intensity, third index.

**"stan"**

Sets the output to standard file format, which is used by KUPL. The output file contains two initial lines and then a block for each data line separated by an empty line. The first line contains the number of data points along the abscissa and the ordinate, the second line the minimum

and maximum value of the indices in reciprocal space along the abscissa and ordinate respectively.

**"shelxl" [, {"fobs" | "fcalc"}]**

Sets the input file format to SHELXL. DISCUS assumes the input file to be a SHLEXL List type 5 containing:

```
<h> <k> 1> <Fobs> <Fcalc> <phase angle in degree>
```

The file is read as free format. All reflections are expanded by the appropriate symmetry operations, including proper phase transformation.

If the optional parameter is "fobs", the observed structure factor is used for the calculation, if the parameter is "fcalc", the calculated structure factor is used.

**"hklf4" [,"all"]**

Sets the input file format to the SHELX HKLF4 format. Without the optional parameter, DISCUS assumes the input file to be a SHLEXL HKLF4 file AFTER merging symmetrically equivalent reflections containing:

```
<h> <k> 1> <Intensity> <sigma>
```

The file is read as free format. All reflections are expanded by the appropriate symmetry operations, including proper phase transformation.

If the optional parameter "all" is given, DISCUS will not expand the reflections by the reciprocal space symmetry operations. This is useful, if you have a data set that represents all of reciprocal space and has not been merged to the asymmetric unit in reciprocal space.

If you use this option, DISCUS automatically assumes that a PATTERSON will be calculated.

## layer

**layer <e11,e12,e13, e21,e22,e23, e31,e32,e33, inc1,inc2>**

sets the lower left, lower right and upper left corner of the Fourier plane to be calculated and sets the number of data points along the horizontal and the vertical

## ll

**ll <e11> , <e12> , <e13>**

Sets the lower left corner of the Fourier plane to be calculated.

## lr

**lr <e21> , <e22> , <e23>**

Sets the lower right corner of the Fourier plane to be calculated.

## na

**na <inc1>**

Sets the number of data points that will be calculated along the abscissa.

**no**

**no** <inc2>

Sets the number of data points that will be calculated along the ordinate.

**ord**

**ord** <switch>

Defines which the component of the vector parallel to the ordinate is written to the output file.

**rhoabs**

**rhoabs** <switch>

Defines which the component of the vector parallel to the real space abscissa is written to the output file.

**rholayer**

**rholayer** <e11,e12,e13, e21,e22,e23, e31,e32,e33, inc1,inc2>

sets the lower left, lower right and upper left corner of the real space plane to be calculated and sets the number of data points along the horizontal and the vertical

**rholl**

**rholl** <e11> , <e12> , <e13>

Sets the lower left corner of the real space plane to be calculated.

**rholr**

**rholr** <e21> , <e22> , <e23>

Sets the lower right corner of the real space plane to be calculated.

**rhona**

**rhona** <inc1>

Sets the number of data points that will be calculated along the real space abscissa.

**rhono**

**rhono** <inc2>

Sets the number of data points that will be calculated along the real space ordinate.

**rhoord****rhoord** <switch>

Defines which the component of the vector parallel to the real space ordinate is written to the output file.

**rhoul****rhoul** <e31> , <e32> , <e33>

Sets the upper left corner of the real space plane to be calculated.

**run****run**

Starts the Patterson calculation.

**scale****scale** <scale-factor>

Defines an overall scale factor. Together with the automatic scaling by 1/vol(unit cell), this scale factor serves to put the electron density on an absolute scale of electrons per cubic Angstrom. The scale factor should effectively multiply the observed F(000) such that it is equal to F(000) calculated for one unit cell of the average structure.

Every time you enter "patterson" the scale factor is calculated to be number of atoms in crystal / number of atoms in one unit cell. If you use experimental data, you MUST specify the scale factor every time you enter the "patterson" submenu.

**set**

```
set "accu",{ "init"|"add" }
set "excl",{ "none" | <value> }
set "mode",{ "normal"|"sharp"|"super" }
set "origin",{ "normal"|"subtract" }
set "rsym",{ "appl"|"igno" }
```

Set defines various parameters.

**accu****"accu"**, { "init"|"add" }

Sets the mode for the accumulation of Patterson transforms.

**"init"**

The electron density array is initialized, i.e. set to zero.

**"add"**



The next Patterson transform is added to the previous value. This way several Patterson transforms will contribute to a single layer in direct space, which is necessary to create a Patterson section rather than a projection onto the layer. See the manual for further details.

If you use DISCUS to calculate the input data for the Patterson, you **MUST** separate the loop that calculates the Fourier from the loop that calculates the Patterson!!! Both parts of the program use the same variable to store the results and by mixing the loops, the result will be unpredictable nonsense !

### **excl**

**set "excl",{"none"<value>}**

Signals whether certain data points should be ignored.

**"none"**

All data points are treated as normal data points, regardless of their value. This is the default at program start.

**<value>**

Data points that are equal to <value> are excluded from the Patterson transform. You can use this option to mask out regions of the input data by setting their values to <value>. In the companion program KUPLOT, data points of value equal to -9999. are treated as excluded regions.

### **mode**

**set "mode",{"normal"|"sharp"|"super"}**

Determines the calculation mode of the Patterson function. This mode may be a: "normal" Patterson using  $|F|^*|F|$  "sharp" Patterson using  $|F|^*|E|$  "super"-sharpened Patterson using  $|E|^*|E|$  Currently the mode can only be used in combination with SHELX HKLF4 files.

### **origin**

**set "origin",{"normal"|"subtract"}**

Determines whether the origin peak should be subtracted from the Patterson function. Since this depends on the data to be on an absolute scale, a Wilson plot is calculated and DISCUS puts the data on an approximately absolute scale.

Currently the origin peak can only be subtracted in combination with SHELX HKLF4 files.

### **rsym**

**set "rsym",{"appl"|"igno"}**

Determines whether the space group symmetry is applied to all reflections. If the second parameter is "appl", all symmetry elements are applied to all reflections and the phases are transformed accordingly. This allows you to read a set of unique reflections and calculate the Patterson.

**show****show**

shows the current settings for the Fourier transform file

**type**

```
type {"a"|"b"},
{"intensity"|"amplitude"|"real"|"imaginary"|"phase"|"fobs"|"fcalc"}
```

Defines the content of file "a" or "b".

The Patterson transform is calculated from one or two input files. The two files must contain the intensity, real or imaginary part, or amplitude in one of the following combinations:

```
file a          file b
-----
intensity
amplitude
real            imaginary
```

If the input format is "shelxl", DISCUS reads just one file. You can calculate the inverse Fourier from Fobs or Fcalc data by choosing the appropriate file type from one of the following choices:

```
file a          file b
-----
amplitude
fobs
fcalc
```

**stat**

```
stat {"screen" | <outputfile> }
```

This command calculates the normalized structure factor, and displays relevant statistical values like  $\langle |E|^2 - 1 \rangle$  for all reflections and several sections in reciprocal space.

It requires the input file to be a SHELX HKLF4 file. The results are written to screen only or optionally into the outputfiles: <outputfile>.ehkl Normalized structure factors for all input reflections <outputfile>.statistics Statistical values for the E <outputfile>.short Short summary of statistical values <outputfile>.histogramm Number of E versus |E| for plotting

The input file must have been defined by `==> "file a,<infile>"`, For the statistics to be correct, the correct unit cell must have been defined.

**ul**

```
ul <e31> , <e32> , <e33>
```

Sets the upper left corner of the Fourier plane to be calculated.

**wilson****wilson**

This command reads a HKLF4 file and calculates Wilson statistics from a plot of  $\ln(\text{average}(\text{formfactor}^2)/\text{average}(\sin(\theta)/\lambda))$  versus  $\sin(\theta)/\lambda$ . The scale factors:  $F_{\text{calc}} = k \cdot F_{\text{obs}}$   $I_{\text{calc}} = k^2 \cdot F_{\text{obs}}$  and an overall B, respectively U are calculated and displayed. A histogram of the logarithms versus  $\sin(\theta)/\lambda$  is written to file `<file>.wilson`, where `<file>` is the base of the HKLF4 filename.

The input file must have been defined by `==> "file a,<infile>"`, The correct radiation and wavelength must have been defined within the `==> 'fourier'` menu, and the structure must contain elements. For the Wilson statistics to be correct, the relative amount of atoms should be equal to or at least close to the composition of the actual substance to which the data set belongs.

**A.31 pdf****pdf**

This command switches to the PDF mode of DISCUS. PDF stands for Pair-Distribution-Function. The function actually used in DISCUS is the so-called reduced radial distribution function  $G(r)$  which is defined as

$$G(r) = 4 \pi r ( \rho(r) - \rho(0) )$$

The value of " $\rho(r)$ " is the probability density of finding an atom at the distance " $r$ " from a given atom. Besides calculating  $G(r)$  from a given structure, the structure can be modified to match a observed  $G(r)$  via the RMC method. Note, that this sublevel used many RMC commands and will overwrite possible settings made in the RMC segment. All commands are listed in the help entry 'commands'. Only commands unique to this level are described below.

**commands**

Valid commands at this sublevel are :

```
@      ! Executes a macro (see main help)
=      ! Algebra (see main help)
```

`calc !` Calculates PDF for current structure

```
data      ! reads observed PDF for RMC refinement
dese      ! Deselects used atomtypes (see RMC level)
continue! Continue a stopped macro (see main help level)
echo      ! Echos a string, just for interactive check
eval      ! Evaluates an expression (see main help)
exit      ! Terminates the PDF sub level
help      ! Gives on-line help for 'pdf' (see main help)
i/jdese   ! Deselect atoms (i/j) for PDF calculation
i/jsele   ! Select atoms (i/j) for PDF calculation
mdes      ! Deselects molecules (see RMC level)
msel      ! Selects molecules for refinement (see RMC level)
reset     ! Reset PDF module
run       ! Start PDF refinement (see RMC level)
save      ! Saves structure or PDF to given file
sele      ! Selects used atomtypes (see RMC level)
```

```
set      ! Sets most PDF parameters
show     ! Show current PDF settings
stop     ! Stops execution of a macro (see main help level)
system   ! Executes operating system command (see main help)
wait     ! Waits for user input (see main help)
```

## calc

### calc

This command will calculate the PDF of the current structure. Note that the result is NOT automatically saved. This must be done using the command 'save' (-> save).

## data

### data <file>

This command will read an observed PDF from the file <file>. The file needs to be ASCII format with the following information on each line: r G dr w(G). Here 'r' is the distance and G the observed value of the reduced radial distribution function G(r). The value 'dr' is ignored and the last value gives the weight w(G) of G(r) to be used in the refinement (e.g. 1/(error of G)). NOTE: The data file must be in constant r steps and the first points MUST be equal to r=stepsize.

## i/jdese

```
idese { "all" | <name> | <number> }, [ ... ]
```

```
jdese { "all" | <name> | <number> }, [ ... ]
```

This command deselects atom types given either by <name> or <number> for the PDF calculation. The two commands allow one to deselect atom types for each atom in a pair 'ij' contributing to the PDF calculation.

## i/jsele

```
isele { "all" | <name> | <number> }, [ ... ]
```

```
jsele { "all" | <name> | <number> }, [ ... ]
```

This command selects atom types given either by <name> or <number> for the PDF calculation. All other atoms are ignored. This allows the calculation of differential or partial PDFs. The two commands allow one to select atom types for each atom in a pair 'ij' contributing to the PDF calculation.

## rese

### rese

This command resets most PDF module settings.

**run****run**

This command starts the RMC refinement to match the calculated PDF with the observed one.

**save**

**save** {"pdf" | "stru" | "mark"}, <file>

This command allows to save either the current PDF ("pdf") or the current structure ("stru") to the file named <file>. The parameter "mark" saves the distances of all pairs in <file> for plotting markers below a PDF. Markers can only be saved if the Gaussian mode is disabled.

**set****set "subcommand"**

This command allows to set most of the PDF parameters. Some of the commands are explained in the RMC section (-> rmc). The following "subcommands" are valid:

**commands**

Valid subcommands are:

```
"bound"      : toggles the use of periodic boundaries
"cycl"       : sets number of cycles to be calculated (see RMC level)
"delta"      : sets quadratic correlation factor (Gaussian mode only)
"density"    : sets rho0 value manually
"diameter"   : sets particle diameter for finite objects
"disp"       : sets output interval (see RMC level)
"finite"     : sets the correction type for 4Pi Rho for finite objects
"frange"     : sets the range in r used for the refinement
"gamma"      : sets linear correlation factor (Gaussian mode only)
"mdis"       : sets minimal allowed distances between atoms (see RMC level)
"mode"       : sets RMC mode (relaxation/switch atoms) (see RMC level)
"move"       : sets sigma for generated RMC shifts (see RMC level)
"partial"    : sets partial weights for PDF (use with MIXSCAT program)
"poly"       : sets the background polynomial for finite objects
"qalp"       : sets resolution broadening (Gaussian mode only)
"qmax"       : sets maximal Q value used for termination correction
"qsig"       : sets SIGMA Q for resolution correction
"rad"        : sets radiation used to calculate PDF
"range"      : sets r range for PDF calculation
"rdensity"   : sets rho0 correction factor manually (see "density")
"scal"       : sets scale factor and refinement flag (see RMC level)
"shape"      : sets particle shape parameter for finite objects
"sigm"       : sets SIGMA for CHI2 calculation (see RMC level)
"srat"       : sets sharpening of low r peaks (Gaussian mode only)
"therm"      : toggles the convolution with thermal Gaussian
"weight"     : sets weight correction factor for finite objects
```

**bound**

```
set "bound","periodic" [,"2D" | "3D"]
set "bound","crystal" [,"quick" | "exact"]
```

This command allows the user to select whether periodic boundaries should be applied or not. In case of "periodic" periodic boundaries will be applied and the optional third parameter will determine whether periodic boundaries are applied in all directions ("3D") or only in those direction of the crystal that extends over more than a single unit cell ("2D"). The default is "3D". Note that in case of periodic boundaries, the calculation mode is always unit cell requiring the proper order of atoms in memory (i.e. no purge). If periodic boundary conditions are used, finite spherical objects can still be simulated by the use of `==> "set finite,sphere,<diameter>".` The parameter "crystal" disables periodic boundaries. In this case the last optional parameters selects the calculation mode. The parameter "quick" will result in only neighboring unit cells being included again relying on the correct order of atoms in memory. The value "exact" will include all atoms making the calculation much slower, but it will work independent of the atom order, e.g. after a 'purge' command or with an arbitrary structure not generated by DISCUS. If periodic boundary conditions are not used, the background line  $-4\pi\rho r$  should be corrected by parameters defined by `==> "set finite".`

### **corrlinear**

**set "corrlinear",<g>**

If the Gaussian convolution mode (`-> set therm`) is used, the PDF peak width will be modified by the term  $-g/r$ . This linear dynamic correlation factor `<g>` is specified with this command. A value of 0.0 will turn off this correction.

Identical to "gamma"

### **corrquadratic**

**set "corrquadratic",<d>**

If the Gaussian convolution mode (`-> set therm`) is used, the PDF peak width will be modified by the term  $-d/r^2$ . The quadratic dynamic correlation factor `<d>` is specified with this command. A value of 0.0 will turn off this correction.

Identical to "delta"

### **delta**

**set "delta",<d>**

If the Gaussian convolution mode (`-> set therm`) is used, the PDF peak width will be modified by the term  $-d/r^2$ . The quadratic dynamic correlation factor `<d>` is specified with this command. A value of 0.0 will turn off this correction.

Identical to "corrquadratic"

### **density**

**set "density", [<rho0> | "auto"]**

**set "rdensity", [<rho0> | "auto"]**

This commands allows to specify a user defined number density `<rho0>`, e.g. in cases where no unit cell is defined. To use the number density determined by the unit cell volume, simply

use the parameter "auto" (default). The command "rdensity" works in the same way, except  $\langle \rho_0 \rangle$  is treated as a factor for the calculated  $\rho_0$  value.

### diameter

**set "diameter", $\langle$ diameter $\rangle$**

This commands allows to specify a user defined particle diameter that is used in combination with the finite size treatment ==> "set finite,tanh"

### finite

**set "finite", {"periodic" | "poly", $\langle$ diameter $\rangle$  | "sphere", $\langle$ diameter $\rangle$  | "tanh"}**

This commands defines how to deal with finite particles. Since their PDF does not contain distances beyond the diameter, the background line  $-4\pi \rho_0 r$  must be corrected.

"periodic" The program is instructed not to apply a correction. The crystal is periodic. Also, use ==> "set bound,periodic".

"poly", $\langle$ diameter $\rangle$  A polynomial function of order N is used to correct the  $-4\pi \rho_0 r$  line. See ==> "set poly" for the corresponding parameters. The  $-4\pi \rho_0 r$  line is set to zero for distances longer than the diameter.

"sphere", $\langle$ diameter $\rangle$

The effect of this setting depend on the periodic boundary flag. If periodic boundaries are chosen, the 'periodic'  $g(r)$  is multiplied by the shape function of a sphere of diameter  $\langle$ diameter $\rangle$ .  $f = 1 - 3/2 r/\text{diameter} + 1/2 (r/\text{diameter})^3$  For  $r > \text{diameter}$   $g(r)$  is set to zero.

If periodic boundaries are disabled, only the  $-4\pi \rho_0 r$  line is multiplied by the shape function of a sphere of diameter  $\langle$ diameter $\rangle$ .  $f = 1 - 3/2 r/\text{diameter} + 1/2 (r/\text{diameter})^3$  For  $r > \text{diameter}$  the background line is set to zero.

"tanh" A tanh function is used to correct the  $-4\pi \rho_0 r$  line. See ==> "set diameter" and "set shape" for the corresponding parameters.

The only settings that may be used if the boundaries are set to periodic, are "periodic" and "sphere".

### frange

**set "frange", $\langle$ rmin $\rangle$ , $\langle$ rmax $\rangle$**

This command sets the range in "r" that will be used for the refinement. The range of the PDF that will be saved using the command 'save' is also determined by these settings.

### gamma

**set "gamma", $\langle$ g $\rangle$**

If the Gaussian convolution mode (-> set therm) is used, the PDF peak width will be modified by the term  $-g/r$ . This linear dynamic correlation factor  $\langle$ g $\rangle$  is specified with this command. A value of 0.0 will turn off this correction.

Identical to "corrlinear"

**partial**

```
set "partial","internal"
set "partial",<a1>,<a2>,<w>
```

This command allows one to specify the weight <w> for each partial from atoms <a1> and <a2>. This can be used in conjunction with the program MIXSCAT to calculate the correctly weighted differential PDFs from a model. If the second parameter is "internal", the weights are calculated from the scattering lengths of the elements (default).

**poly**

```
set "poly",<p1> [,<p2>...]
```

This command allows to set the parameters of a polynomial function that is used in combination with the finite size treatment ==>"set finite,poly". The background line  $-4 \pi \rho r$  is modified by subtracting this polynomial:  $-4 \pi \rho r - p_1 r^1 - p_2 r^2 - p_3 r^3 - p_4 r^4 - p_5 r^5$ . The order of the polynomial is defined by the number of parameters given, which can be at most equal to five.

Notice that the polynomial starts with the linear term and does not contain a term  $p_0 r^0$  !

**qalp**

```
set "qalpha",<alpha>
```

PDF peaks are broadened at large values of  $r$  due to the instrument resolution of the measurement. This command sets the parameter <alp> controlling the broadening which has the functional form  $\langle \text{alp} \rangle^2 \cdot r^2$ . Identical to "qbroad"

**qbroad**

```
set "qbroad",<qbroad>
```

PDF peaks are broadened at large values of  $r$  due to the instrument resolution of the measurement. This command sets the parameter <qbroad> controlling the broadening which has the functional form  $\langle \text{qbroad} \rangle^2 \cdot r^2$ . Identical to "qalpha"

**qmax**

```
set "qmax",<qmax>
```

Since we can only measure the scattering up to a maximum value in  $Q$ , the resulting PDF will show termination ripples. In order to account for these, termination ripples can be calculated for the computed PDF. In order to do this the actual maximum  $Q$  value <qmax> in  $\text{\AA}^{-1}$  must be entered. If the value is zero NO termination correction will be applied.

**qdamp**

```
set "qdamp",<qdamp>
```



This allows to enter a value `<qdamp>` for the correction of the limited resolution of the experiment. This results in a dampening of the  $G(r)$  function with increasing values of  $r$ . Again if the entered value is zero, no correction will be applied. Identical to "qsigma"

### qsigma

**set "qsigma",`<qsig>`**

This allows to enter a value `<qsig>` for the correction of the limited resolution of the experiment. This results in a dampening of the  $G(r)$  function with increasing values of  $r$ . Again if the entered value is zero, no correction will be applied. Identical to "qdamp"

### rad

**set "rad","neutron"**

**set "rad","xray" [,`<qx>`]**

**set "rad","electron" [,`<qx>`]**

This command selects the radiation used to calculate the PDF. The weighting factor for each atom pair is  $B(i) \cdot B(j) / \langle B^2 \rangle$ . In case of neutrons ("neutron"), the values  $B(i)$  are simply the scattering length of the individual elements. For Xrays ("xray"), or electrons ("electron") the situation is more complicated since the scattering power varies as a function of  $Q$ . As an approximation the scattering power at a give  $Q$  value `<qx>` in  $\text{\AA}^{-1}$  is used. The value `<qx>` can be taken as zero (default) in which case the number of electrons determines the weight. One other common way is to use the  $Q$  of the first peak in the diffraction pattern.

### range

**set "range",`<rmax>`,`<dr>`**

This command allows to set the maximum  $r$  and the step width used to calculate the PDF. Note that if observed data are used, these values are set to the corresponding values of the actual data file. All values are entered in units of  $\text{\AA}$ .

### shape

**set "shape",`<shape>`**

This commands allows to specify a user defined particle shape parameter that is used in combination with the finite size treatment `==>"set finite,tanh"`

### srat

**set "srat",`<rat>`,`<rcut>`**

This command allows the user to sharpen PDF peaks below a value of  $r = \text{rcut}$  by a factor of `<srat>` (usually  $< 1.0$ ). Note that this will effect only calculations using the Gaussian mode (`-> set therm`).

**therm**

**set "therm", {"off" | "gauss"}**

This command allows the user to select between two calculation modes. If the parameter "gauss" is given, the contribution of each atom pair to the PDF is convoluted with a Gaussian. The width of this Gaussian is given by the individual thermal parameters B. If "off" is used, the contribution is a single "delta" type function.

**weight**

**set weight,<weight>**

Allows you to correct the weight factor used to multiply  $g(r)$  before adding it to  $-4\pi \rho_0 r$ . For infinite objects this should be left at the default value of one. For finite objects or in case of uncertainties in the chemical composition it may differ.

**show**

**show [{"all" | "pdf" | "mode" | "atom"}]**

This command shows the various settings. If the command is entered without further parameters, the PDF settings are shown similar to the situation when the parameter is "pdf". The parameter "mode" results in the output of the current RMC mode. Note that some settings are listed that are not used by the PDF RMC refinement. The parameter "atom" corresponds to the listing of parameters like selected atoms, sizes of moves and minimal allowed distances between atom types. Finally the parameter "all" will cause all settings to be printed on the screen.

**A.32 plot****plot**

Switches to the structure plotting submenu. The structure can be written in a format suitable for plotting by several structure plotting programs such as KUPPL, GNUPLOT.

Further help topics are:

**commands**

Valid commands at this sub level are:

```
@      ! Executes a macro (see main help)
=      ! assigns the value to a variable (see main help)
absc   ! Defines the abscissa of the plot
asym   ! Lists the content of the asymmetric unit
chem   ! Lists all type of atoms present in the crystal
col    ! Defines the sequence in which the xy and z coordinates are written.
continue! Coninue a stopped macro (see main help level)
des    ! Deselects atoms
echo   ! Echos a string, just for interactive check (see main help)
exit   ! Terminates the plot sub level, returns to the main DISCUS level.
ext    ! Sets the extend of crystal space to be plotted
```

```

help      ! Gives on-line help for 'plot' (see main help)
hkl       ! Sets reciprocal space direction normal to plot slice
ordi      ! Defines the ordinate of the plot
outfile   ! Name of output file to which the structure is written
prog      ! Selects the plotting program
prop      ! Selects the atom properties, which will be checked
run        ! Starts writing of the structure
sel        ! Selects which atoms are to be included in the plot
set        ! Sets type, color and size of atom in plot file
show      ! Shows settings for the plotting parameters
thick     ! Sets the half thickness of the plot slice
uvw       ! Sets direct space direction normal to plot slice
stop       ! Stops execution of a macro (see main help level)
system    ! Executes operating system command (see main help)
type      ! Sets type of output - crystal or unit cell projection
wait      ! Waits for user input (see main help)
vec       ! Sets a point inside plot slice

```

## absc

**abscissa** <x>,<y>,<z>

Defines the abscissa of the plot. Instead of using the atom coordinates with respect to the crystallographic axes, the atoms are transformed into the system based on the abscissa, the ==> 'ordinate' and the normal to the plot slice ==> 'uvw', 'hkl'. Use these commands to plot the projection onto any plain other than the xy-, xz- or yz-plane.

## asym

**asym**

Shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the variable m[<index>]. If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

## bonds

**bonds** { "all" | <name> | <number> } , { "all" | <name> | <number> } ,  
<rmin> , <rmax> , <diameter> , { <gray> | <r> , <g> , <b> }

Defines which bonds the plotting program XBS will draw. This command does not affect the other plotting programs. Choose the pairs of atoms to be connected by bonds. Bonds that fall into the intervall <rmin> to <rmax> are plotted. The thickness of the bond is set to <diameter>. You can specify the bond color as either a gray value in the range [0 - 1] or an RGB triplet.

**chem****chem**

Shows the type of all atoms present in the crystal. For each different atom present, its identifying number, its name and its temperature coefficient are listed. The list contains all the atoms in the asymmetric unit plus any atoms added to the structure. If atoms are removed from the structure, the program does not check whether there are any atoms of this type left. If a particular type of atom is completely removed from the structure, it will remain in the list of different atoms, and will be displayed by 'chem'.

**col**

**col** { "xyz" | "yzx" | "zxy" | "zyx" | "yxz" | "xzy" }

Defines the sequence in which the xy and z coordinates of the atoms are written. If the 'prog' "gnuplot" is used, this will allow the user to view the structure from different directions. For "kupi" this will give projections of the structure along the last direction.

If you plot the projection along a vector selected by ==> 'uvw', 'hkl' the interpretation of the coordinate triplets takes a different meaning: x represents the coordinate along the ==> abscissa y represents the coordinate along the ==> ordinate z represents the coordinate along the ==> 'uvw' direction. The coordinates are, in this case, written as: 1.coordinate 2.coordinate 3.coordinate xyz abscissa ordinate uvw zxy uvw abscissa ordinate yzx ordinate uvw abscissa yxz ordinate abscissa uvw xzy abscissa uvw ordinate zyx uvw ordinate abscissa

For "xyz" the plot will show the projection along uvw onto the plane of the abscissa and ordinate. For "zxy" the plot will show the projection along the ordinate onto the plane of the uvw-direction and the abscissa. etc.

Since the thickness of the slice will often be limited by the ==>'thick' command, the sequence 'xyz' would be the most useful one.

**des**

**dese** { "all" | <name> | <number> } [, <name> | <number> } ... ]

Deselects the atoms.

```
"all"      All atoms are deselected, no atom will be plotted.
<name>     Only the atom specified by <name> is deselected.
<number>   Only the atom specified by the scattering type <number> is
           deselected.
```

More than one atom can be deselected by the command.

**ext**

**ext** { "all" | "x",<xmin>,<xmax> | "y",<ymin>,<ymax> | "z",<zmin>,<zmax> }

Sets the extend of crystal space to be plotted.

```
"all"      The whole crystal is written to the plot file.
"x"        The extend along the x direction is limited to (including)
           <xmin> and <xmax>.
```

```
"y"    The extend along the y direction is limited to (including)
        <ymin> and <ymax>.
"z"    The extend along the z direction is limited to (including)
        <zmin> and <zmax>.
```

If any of the parameters "x","y","z" is used, the extend along the other directions is not changed, even if previously set with the "all" command.

## hkl

**hkl** <h>,<k>,<l>

Sets the reciprocal space direction normal to the plot slice. the corresponding direct space direction ==> 'uvw' is calculated automatically. If <hkl> is set to "000", all atoms are plotted, otherwise all atoms are written to the output file, if they are within a slice normal to the <hkl> (and <uvw>) direction. The thickness of the plot slice is defined by the ==> 'thick' command, while the ==> 'vec' command defines a point that represents the center of the slice. Further restrictions on the atoms to be included can be applied by the ==> 'sel' and the ==> 'exte' commands.

## mdes

**mdes** { "all" | <number> } [, <number> ...]

This command deselects all or individual molecule types used for the plotting output.

## mole

**mole** { "all" | "origin" }

This command allows the user to specify whether all ("all") atoms of a molecule should be used in the plot or only the origin ("origin") of the molecule should be used.

## msel

**msel** { "all" | <number> } [, <number> ...]

This command allows to select the molecule types to be used for the plotting output. By using 'msel' the atom selection becomes invalid. The parameter "all" will select ALL molecule types. Alternatively individual molecule types can be selected using the corresponding <number>. In order to use atoms again, use the 'sele' command.

## ordi

**ordinate** <x>,<y>,<z>

Defines the ordinate of the plot. Instead of using the atom coordinates with respect to the crystallographic axes, the atoms are transformed into the system based on the ==> 'abscissa', 'ordinate' and the normal to the plot slice ==> 'uvw', 'hkl'. Use these commands to plot the projection onto any plain other than the xy-, xz- or yz-plane.

**outfile****outfile** <filename>

Sets the name of the output file to which the structure is written.

**prog**

```
prog { "gnuplot" | "kuplot" | "atoms" |
      "xbs" | "frames" [, "append" | "init"] |
      "diamond" | "jmol" | "drawxtl" |
      "cif" }
```

Selects the plotting program for which the output is to be formatted.

**"gnuplot"** The xyz coordinates of all selected atoms are written in the sequence defined by the ==> 'col' command.

**"kuplot"** The xyz coordinates of all selected atoms are written in the sequence defined by the ==> 'col' command. A projection of the structure along the third direction of the 'col' parameter is plotted. The file contains the typ, color and size code set with the ==> 'set' command. Use 'load cr,<file>' to read the file with KUPLOT.

**"atoms"** The coordinates are written in a format that can be imported in ATOMS using the 'import - free format' option. In order to trick ATOMS into displaying the complete crystal, a new unit cell with the size of the crystal is used and the space group is set to P1.

**"xbs"** The coordinates are written in the input format for XBS. All atoms are automatically transformed into cartesian space. Cartesian b axis is parallel to the crystal b axis, the cartesian c axis is parallel to the crystal c\* axis, and the cartesian a axis is normal to b and c\*, righthanded. The atom color are set with ==> 'set', the bonds that will be plotted are set with ==> 'bonds'. The scale is set automatically, the initial viewing direction is along the ==> 'uvw' or 'hkl', the initial abszissa is along ==> 'abs'. The abszissa you specify is projected into the plane normal to the 'uvw' direction. You do not have to worry about the angle, as long as it is not equal to zero degrees.

**"frames"** A movie file in the format of XBS is written to the output file. Remember to change the output filename to consist of an identical base as had been used for the main XBS output. If the optional second parameter is "init", the frame file is initialized. If the second parameter is "append", or if no second parameter is given, the current atom positions are added to the existing file. If the file does not exist, it is created.

**"diamond"** The coordinates are written as the DIAMOND XYZ file type. space. Cartesian b axis is parallel to the crystal b axis, the cartesian c axis is parallel to the crystal c\* axis, and the cartesian a axis is normal to b and c\*, righthanded. All atoms are automatically transformed into cartesian space. Import the file into DIAMOND by opening the file under the 'file' menu. Use file type 'XYZ'. Display all atoms using the "Add all Atoms" switch in the "Structure" menu. Caution, if only a partial crystal is written, the number of atoms is currently wrong. Use a "cif" format instead.

"jmol"      The coordinates are written as the JMOL XYZ file type. space. Cartesian b axis is parallel to the crystal b axis, the cartesian c axis is parallel to the crystal c\* axis, and the cartesian a axis is normal to b and c\*, righthanded. All atoms are automatically transformed into cartesian space. Import the file into JMOL by opening the file under the 'file' menu. Use file type 'XYZ'. Display all atoms using the "Add all Atoms" switch in the "Structure" menu.  
 Caution, if only a partial crystal is written, the number of atoms is currently wrong. Use a "cif" format instead.

"drawxtl"      The atoms are written in a format to be read by DRAWXTL.

"cif"      This creates a CIF file from the selected atoms. All coordinates are converted to a single unit cell.

Use this output format for Atoms, Diamond, or JMOL as well.

## prop

**prop** { "ignore" | "present" | "absent" }, <property> [, <property> ...]

Defines which properties an atom must have to be included in the plot.

"ignore" The properties listed will be ignored. An atom will be included in the plot regardless, whether it has the properties or not, "present" An atom must have the properties listed, to be included in the plot "absent" An atom is not allowed to have the properties listed.

The properties can be one of: "all" Apply the rule to all properties. "normal" The rules apply if an atom is a normal atom (not a void) "molecule" The rules apply if an atom is part of a molecule "domain" The rules apply if an atom is part of a domain "outside" The rules apply if an atom is outside of the crystal. This status is set if the atom has been cut off by a ==> 'boundary' command. "external" The rules apply if an atom is close to an external surface, either on the inside or on the outside. The flag is set by the ==> 'boundary' command. Individual distances to the boundary can be set within the ==> 'surface' menu through the ==> 'surface/fuzzy' command. "internal" The rules apply if an atom is close to an internal surface, either within the host or within the guest structure. The flag is set by placing a guest structure into the crystal through the ==> 'domain' menu. Individual distances to the boundary can be set within the ==> 'domain' menu through the ==> 'domain/fuzzy' command.

## run

### run

Starts writing of the structure. If no output filename has been provided or no atoms been selected, an error message is given.

## sel

**sel** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Defines which atoms are included in the output. Possible values for the first mandatory parameter are mutually exclusively:

"all"      all atoms of the crystal are included.

This includes the "voids" in the structure, which are stored as scattering curve number zero.

```
<name>    all the atoms called <name> of the crystal are included.
           This includes symmetrically not equivalent atoms.
<number>  all atoms of the crystal that are of scattering type <number>
           are included.
```

The selection made stay valid until explicitly deselected!

## set

```
set { <name> | <number> }, <typ>, <col>, <size>
set { <name> | <number> }, <typ>, <r>, <g>, <b>, <size>
```

This command allows to set the marker type <typ>, its color <col> and size <size> for the given atom or molecule. In case of atoms, the input can be given by its name <name> or scattering type number <number>. For molecules only <number> as the molecule type is a valid input. The resulting file will contains the coordinates of the atom and typ, color and size in a row. This file can be read by KUPLOT using the command

```
load cr,<file>
```

The possible values for <typ> are:

0 : no marker	5 : filled square	10 : line (/)
1 : dot	6 : empty triangle	11 : line (\)
2 : empty circle	7 : cross (x)	12 : line (-)
3 : filled circle	8 : cross (+)	13 : vertical line
4 : empty square	9 : line ( )	from y-axis

The possible values for the colors <col> are:

The allowed colour values <icol> are:

1 : red	5 : yellow	9 : dark blue	13 : cyan
2 : green	6 : black	10 : dark magenta	14 : dark cyan
3 : blue	7 : dark red	11 : dark yellow	15 : white (!)
4 : magenta	8 : dark green	12 : gray	

The size of the markers <size> is given relative to the marker size set in the plot program KUPLOT, i.e. a value of 1.0 will result in the full marker size whereas 0.5 will reduce the size by 50%.

If you are using 'gnuplot' as program typ (-> 'prog') the settings will be ignored.

If you are using the XBS program, the color can be represented as an RGB triplet. If you give one value only, it is interpreted as gray shade in the range [0-1].

## show

### show

Shows the settings for the plotting parameters.



**thick****thic** <value>

Sets the half thickness of the plot slice in Angstrom. All atoms are written to the output file, if they are within a slice of thickness  $2 \times \text{<value>}$  that contains the direct space point defined by  $\Rightarrow$  'vec' and is normal to the direction defined by  $\Rightarrow$  'hkl' or 'uvw'. Further restrictions on the atoms to be included can be applied by the  $\Rightarrow$  'sel' and the  $\Rightarrow$  'exte' commands.

**title****title** <title>

A string is read, which is used as title for the current frame of an XBS  $\Rightarrow$  'prog frame' frame series.

**uvw****uvw** <u>,<v>,<w>

Sets the direct space direction normal to the plot slice. The corresponding reciprocal space direction <hkl> is calculated automatically. If <hkl> is set to "000", all atoms are plotted, otherwise all atoms are written to the output file, if they are within a slice normal to the <uvw> (and <hkl>) direction. The thickness of the plot slice is defined by the  $\Rightarrow$  'thick' command, while the  $\Rightarrow$  'vec' command defines a point that represents the center of the slice. Further restrictions on the atoms to be included can be applied by the  $\Rightarrow$  'sel' and the  $\Rightarrow$  'exte' commands.

**type****typ** { "crystal" | "projection" }

This command switches between to plot modes. The typ "crystal" will write the atom coordinates of the selected ( $\rightarrow$  'sele') atoms on a file. The type "projection" will remove the translation and all atoms will be projected in a single unit cell. The second type can be read with KUPLOT using 'load cr,<file>' or 'load de,<file>,<dx>,<dy>'. See the KUPLOT online help for more details on the different file formats.

**vector****vec** <v1>,<v1>,<v3>

Sets the vector that represents the center of the plot slice. All atoms are written to the output file, if they are within a slice that is normal to the <hkl> (and <uvw>) direction and contains the direct space vector <v1, v2, v3>. The thickness of the plot slice is defined by the  $\Rightarrow$  'thick' command, while the  $\Rightarrow$  'vec' command defines a point that represents the center of the slice. Further restrictions on the atoms to be included can be applied by the  $\Rightarrow$  'sel' and the  $\Rightarrow$  'exte' commands.

## A.33 powder

Switches to the powder diffraction sublevel of DISCUS.

Two modes are used, the Debye algorithm and a complete integration of reciprocal space.

The Debye mode is best suited for nanoparticles, as size effects are automatically taken into account. It becomes quite slow for large crystals.

The complete integration is best suited for large crystals, especially if large sections of reciprocal space are free from diffuse scattering.

### commands

Valid commands at this sub level are:

```
@      ! Executes a macro (see main help)
=      ! assigns the value to a variable (see main help)
asym   ! Lists the content of the asymmetric unit
chem   ! Lists all type of atoms present in the crystal
continue! Coninue a stopped macro (see main help level)
echo   ! Echos a string, just for interactive check (see main help)
exit   ! Terminates the plot sub level, returns to the main DISCUS level.
help   ! Gives on-line help for 'plot' (see main help)
electron! switches to electron calculation
neut   ! switches to neutron calculation
run     ! Starts writing of the structure
set     ! Sets Theta limits, grid in reciprocal space, ...
show   ! Shows settings for the plotting parameters
stop    ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
wait    ! Waits for user input (see main help)
xray    ! switches to electron calculation
```

### electron

#### electron

switches to electron calculation

### neut

#### neut

Switches to neutron diffraction.

### run

#### run

Starts the calculation of the powder diffraction pattern.

### show

#### show

Shows the current settings relevant to the calculation of a powder pattern.

**set**

Sets various parameters for powder diffraction.

**axis**

**set axis, {"q"|"tth"}**

Defines the units of the x-axis you intend to use. "q" The units are in  $q = 2\pi h = 2\pi d^* = 2\pi 2 \sin(\theta)/\lambda$  "tth" The units are in  $2\theta$

The internal calculations will use the corresponding limits and increments. Thus you need to match: <qmin>, <qmax>, <deltaq> to a "q" axis <tthmin>, <tthmax>, <delatth> to a "tth" axis.

**back**

**set back, <p0> [, <p1> [,...]]**

Set paramaters for a background polynomial.

**bragg**

**set bragg, {"incl"|"excl"}**

This command applies to the full integration mode only. It allows you to include (=default) the Intensities of the Bragg reflections or to exclude these. Actually rarely used these days...

**calc**

**set calc, {"compl"|"debye"}**

Defines the calculation mode. By selecting "compl" DISCUS will calculate the Fourier transform on a complete reciprocal lattice defined by ==> 'dh', 'dk', 'dl'. Alternatively, the switch "debye" will tell DISCUS to calculate the powder pattern by using the Debye formula. For details see the user manual.

**delta**

**set delta, <res>**

This command allows to specify the instrument resolution <res>. The calculated powder pattern is convoluted with a Gaussian of the width of <res>. If <res> is set to zero, no convolution is carried out.

See ==> 'set profile' for a convolution by full profile function.

**disp**

**set disp, [<switch>]**

switches anomalous dispersion on /off. If <switch> is "anom" anomalous dispersion will be calculated, if the switch is omitted, no anomalous dispersion is calculated.

**dtth****set dtth,<value>**

The powder pattern is printed at 2-Theta steps of <value>. All reciprocal points are binned to a grid along 2-Theta with intervals of <value>

**dq****set dq,<value>**

The powder pattern is printed at q steps of <value>. All reciprocal points are binned to a grid along q with intervals of <value>

**dh****set dh,<value>**

The calculation is carried out in reciprocal space intervals of <value> long the h00 direction.

**dk****set dk,<value>**

The calculation is carried out in reciprocal space intervals of <value> long the 0k0 direction.

**dl****set dl,<value>**

The calculation is carried out in reciprocal space intervals of <value> long the 00l direction.

**four****set four,{"four"|"stack"}**

Sets the switch for the Fourier calculation mode. For all normal crystals this should be set to "four". If you created a crystal using the ==>'stack' menu, you can speed up The Fourier calculation by setting the switch to "stack". See the stacking fault information regarding the details of this Fourier mode.

**lpcor****set lpcor, "none"****set lpcor, "bragg" [,<theta\_mono>]****set lpcor, "neutron"****set lpcor, "synchrotron" ,<polarisation\_fraction> [,<theta\_mono>]**

This command defines how the powder menu of DISCUS calculates the Lorentz-Polarisation correction.

"none" No correction is calculated

"bragg" [,<theta\_mono>] Bragg-Brentano powder diffraction geometry is set. Optionally you can specify the monochromator angle <theta\_mono>.

"neutron" A neutron powder diffractometer in Debye-Scherrer geometry is used.

"synchrotron" ,<polarisation\_fraction> [,<theta\_mono>] A Debye-Scherrer geometry at a synchrotron source is defined. The polarisation fraction is required. Optionally you can specify the monochromator angle.

### preferred

**set preferred, "off"**

**set preferred, {"rietveld"|"march"}**

**set preferred, {"g1"|"damping"}, <damping>**

**set preferred, {"g2"|"portion"}, <portion>**

**set preferred, {"hkl"}, <h>,<k>,<l>**

This command enables to calculate preferred orientation effects. Currently they can only be used if the full integration of reciprocal space is used, see ==> 'set calc'

"rietveld" Preferred orientation effects are calculated according to the equation:  $P(h) = \text{portion} + (1 - \text{portion}) \cdot \exp(\text{damping} \cdot \alpha^2)$

"march" Preferred orientation effects are calculated according to the modified March equation:  $P(h) = \text{portion} + (1 - \text{portion}) \cdot ((\text{damping} \cdot \cos(\alpha))^2 + \sin^2(\alpha) / \text{damping})^{3/2}$

"portion" or "g1" is the fraction of the powder that is not affected by preferred orientation

"damping" or "g2" is a factor that determines how close a section of reciprocal space has to be oriented to the preferred axis.

"hkl" Defines the preferred orientation axis in reciprocal space

$\alpha^2$  is the square of the acute angle between the vector h and the preferred orientation axis  
hkl.  $\alpha$  is measured in degrees

### profile

**set profile, {"off" | "gauss" | "pseudo"}**

**set profile, "eta", <eta>**

**set profile, "uvw", <u>,<v>,<w>**

**set profile, "asym", <a1>,<b1>,<a2>,<b2>**

**set profile, "width", <width>**

Defines the instrumental profile function.

The powder pattern is convoluted by this profile function.

"off" No convolution by an instrumental profile function is performed. A side effect, the parameter of ==> 'set delta' is set to zero.

"gauss" A simple Gaussian function is used as profile function with sigma as set by the ==> 'set delta' command.

"pseudo" A Pseudo-Voigt function defined as  $\text{eta} \cdot \text{Lorentzian} + (1 - \text{eta}) \cdot \text{Gaussian}$  is used.

"eta" Defines the mixing parameter ETA for the Pseudo-Voigt function

"uvw" Defines the FWHM for the Pseudo-Voigt function. The FWHM is calculated as  $\text{FWHM} = \sqrt{u \cdot \tan^2(2\theta) + v \cdot \tan(2\theta) + w}$

"asym" The Asymmetry or the Pseudo-Voigt function is calculated according to the definition as in Fullprof.

"width" Defines the width in multiples of the FWHM, over which the profile function is applied. The <width> defaults to 20\*FWHM. This is a reasonable compromise between cut of errors at the tail of a peak and computational time.

### scale

#### set scale,<value>

Sets a scale factor. The calculated powder pattern will be multiplied by this scale factor. Most useful in combination with the background mode ==> 'set back'

### sh

#### set sh,<shift\_h>

This command applies to the full integration mode only. It sets a shift of the grid on which intensities are calculated by <shift\_h>. This will allow a substantial gain in computational speed for situations in which only a part of reciprocal space has to be integrated. If, for example, rods of diffuse intensity have to be calculated that have a finite width in reciprocal space in the HK0 plane, then repeated calculations of the powder pattern with shift\_h = plus/minus small values can be used to cover this finite width, while dh can be set to 1.0

### sk

#### set sk,<shift\_k>

This command applies to the full integration mode only. It sets a shift of the grid on which intensities are calculated by <shift\_k>.

See ==> 'set sh' for further info.

### sl

#### set sl,<shift\_l>

This command applies to the full integration mode only. It sets a shift of the grid on which intensities are calculated by <shift\_l>.

See ==> 'set sh' for further info.

### stepr

#### set stepr,<value>

This command applies to the Debye mode only. The interatomic distances are binned to a histogram with stepwidth <value>. This defaults to 0.01 Angstroem at program start. As the histogram array is of fixed size, this limits the maximum distance that debye can handle. See ==> 'show config' at the main DISCUS level for the Debye histogram size. The maximum interatomic distance is then Histogram size \* stepr

**qmin****set qmin,<value>**

Sets the lower limit for q. This must be set, if you intend to use the "Q" axis. You also need to set a 2Theta-range if you want to write the output on a 2Theta scale.

**qmax****set qmax,<value>**

Sets the upper limit for q. The powder pattern is calculated up to this q value. This must be set, if you intend to use the "Q" axis. You also need to set a 2Theta-range if you want to write the output on a 2Theta scale.

**tthmin****set tthmin,<value>**

Sets the lower limit for 2-Theta. This must be set, if you intend to use the "2Theta" axis. You also need to set a Q-range if you want to write the output on a 2Theta scale.

**tthmax****set tthmax,<value>**

Sets the upper limit for 2-Theta. The powder pattern is calculated up to this 2-Theta value. This must be set, if you intend to use the "2Theta" axis. You also need to set a Q-range if you want to write the output on a 2Theta scale.

**temp****set temp,<switch>**

sets whether temperature coefficients are to be used or not. Valid values for <switch> are "use" and "ignore". The first two letters of the switch are sufficient.

**wvle****set wvle,{ <name> | <value> }**

Defines the wave length to be used. Anomalous dispersion will only be calculated if the wave length is given as symbol.

**xray****xray**

Switches to X-ray calculation.

## A.34 project

**project** <u1>,<v1>,<w1>,<u2>,<v2>,<w2> [ ,<flag> ]

This command calculates the projection of vector <uvw 1> onto the second vector <uvw 2> and the projection of the first vector onto the plane normal to the second vector. The <flag> is a string of four characters that indicates whether the input vectors or the output vectors are given in direct or reciprocal space coordinates.

```
The four characters correspond to:
    first input vector,
    second input vector,
    projection of vector 1 onto vector 2,
    projection of vector 1 onto the plane normal to vector 2.
"drdd" means: vector one is given in direct space coordinates, vector 2
               is in reciprocal space coordinates (hkl) and the two
               projections are to be given in direct space coordinates.
Any combination of "d" and "r" is allowed.
The two output vectors are stored in the result variable:
"res[1]","res[2]","res[3]" projection of vector 1 onto vector 2
"res[4]","res[5]","res[6]" projection of vector 1 onto plane normal to
                           vector 2
"res[7]"                  angle between vector 1 and vector 2
```

## A.35 property

### property

Switches to the property menu of DISCUS. Here you can perform operations related to the properties of atoms. Such properties can be: N = normal, the atom is a normal atom (instead of a void) M = molecule, the atom is part of a molecule D = domain, the atom is part of a domain O = outside, the atom is outside of the crystal E = external, the atom is close to an external surface I = internal, the atom is close to an internal surface

The atom properties are set at: reading a unit cell or structure file inserting a domain into a crystal creating an external surface

The atom properties are used when atoms are selected. The global property selection rules apply to commands: 'replace', 'mmc', 'find env' The respective operations for these commands are carried out only if the selected atom fulfils the property selection criteria.

Local property selection rules are used in the menus: 'plot', To apply these, use the local 'property' command within the menu. The local property selection criteria are valid within the menu only.

Further help topics are:

### commands

```
property ! Define how command like replace, remove use the property
set/clear ! Set or clear flag for atoms
show      ! Show current property related settings
```



**prop**

**prop** {"ignore"|"present"|"absent"}, <property> [, <property> ...]

Defines which properties an atom must have to be included in commands ==> 'remove' ; 'replace'

"ignore" The properties listed will be ignored. An atom will be included in the plot regardless, whether it has the properties or not, "present" An atom must have the properties listed, to be included in the plot "absent" An atom is not allowed to have the properties listed.

The properties can be one of: "all" Apply the rule to all properties. "normal" The rules apply if an atom is a normal atom (not a void) "molecule" The rules apply if an atom is part of a molecule "domain" The rules apply if an atom is part of a domain "outside" The rules apply if an atom is outside of the crystal. This status is set if the atom has been cut off by a ==> 'boundary' command. "external" The rules apply if an atom is close to an external surface, either on the inside or on the outside. The flag is set by the ==> 'boundary' command. Individual distances to the boundary can be set within the ==> 'surface' menu through the ==> 'surface/fuzzy' command. "internal" The rules apply if an atom is close to an internal surface, either within the host or within the guest structure. The flag is set by placing a guest structure into the crystal through the ==> 'domain' menu. Individual distances to the boundary can be set within the ==> 'domain' menu through the ==> 'domain/fuzzy' command.

**A.36 set/clear**

**set** <flag>, "types", {"all" | <iscat> [, <iscat...>]}

**set** <flag>, "atoms", {"all" | <start>, <end>}

**clear** <flag>, "types", {"all" | <iscat> [, <iscat...>]}

**clear** <flag>, "atoms", {"all" | <start>, <end>}

Set or clear the property bit for atoms of selected type or for a range of atom numbers.

The flag can be any of "domain", "outside", "external", "internal"

**A.37 purge****purge**

deletes all vacancies in the structure array, thus speeds up Fourier calculation. Due to internal calls, the result variables are changed!

**A.38 r2d**

**r2d** <h>, <k>, <l>

This command transforms reciprocal space vector <hkl> into its corresponding coordinates in direct space. See "r2d" for the reverse transformation.

The resulting vector is stored in "res[1]", "res[2]", "res[3]".

The parallel, equivalent vector, normalized to 1/(length of [h,k,l]) is stored in "res[4]", "res[5]", "res[6]"

## A.39 read

### read

Branches to reading level of DISCUS. At this level a new unit cell or structure may be read. The old structure is lost ! If necessary save it with the `-> "save"` command.

Further help topics are:

### commands

Valid commands at this level:

```
@      ! Execute a macro file (see main help)
cell   ! reads a unit cell from file
lcell  ! reads a unit cell from file (large cells)
echo   ! Echo a string (see main help)
exit   ! Exits read sublevel
free   ! Allow free modification of a structure.
help   ! help to read (see main help)
stru   ! reads a file containing a whole crystal
wait   ! waits for user input (see main help)
```

### cell

**cell** <name> [,<nx> [,<ny> [,<nz>]]]

Reads the contents of a single asymmetric unit from file <name> and then generates the unit cell. Optionally the number of unit cells along x,y,z can be given to generate the whole crystal. The number of unit cells default to 1. The unit cells are arranged symmetrically around the origin, i.e. from  $-nx/2$  to  $+nx/2$  etc. After the unit cell has been read the program returns to the main DISCUS level.

### lcell

**lcell** <name> [,<nx> [,<ny> [,<nz>]]]

This command is similar to `-> cell`, however in contrast to 'cell' only those atoms are treated as new atom type that have a different name or B value. This allows one to read large asymmetric units (e.g. from a protein) without creating that many different atom types.

### free

**free** [<a>,<b>,<c>,<alpha>,<beta>,<gamma>] i [,<name> | <number>]

No structure is read. The space group is set to "P1". The lattice constants are set to the optional values on the command line. Default values are 1,1,1 Angstrom and 90,90,90 degrees. The structure does not contain any atoms.

The optional 7th parameter allows you to set a space group for this empty structure. The parameter may either be the space group name or the space group number. This has no effect on the current structure or on atoms you may insert later. It does allow you to save this structure and then to read it as a unit cell file. It will then be expanded to a full unit cell content according to the space group

**stru****stru** <name>

reads a whole structure from file <name>. The space group information is ignored, nor is a check for consistency of space group and lattice constants performed.

**A.40 remove****remove** <start> [, <end> ]**remove** <start> [, <end> ], "mole" [, "type", <t\_start> [, <t\_end>]]

The parameters <start> and <t\_start> can take one of the following values: { <start> | "all" | "last" }

Removes atom(s) number <start> to <end> from the structure.

If you use the keyword "all", all atoms are removed. If you use the keyword "last", only the last atom is removed.

Internally, this command only sets the scattering type of these atom(s) to zero. Otherwise the atom(s) is treated as if present. Specifically, in the array containing the structure the sequence of atoms is not changed. For this command variation, the atoms are removed, no matter what their individual properties are and whether they fulfill the global property selection rules ==> 'property'. For all atoms that are removed, the property NORMAL is turned off.

The second form of the command removes atoms that are part of the molecule(s) <start> through <end>. Optionally you can specify which molecule type will be removed. Only those molecules of a type in the range <t\_start> through <t\_end> will be removed. The default for this option is all type of molecules.

If you use the keyword "all", all molecule types are removed. If you use the keyword "last", only the last molecule type is removed.

For all atoms that are removed, the property NORMAL is turned off.

**A.41 replace****replace** <index>,<at> [,<dbw>]**replace** <at1>,<at2>,{ "all" | <site> },<prob>**replace** <mo1>,<mo2>,"mol"**replace** <mtyp>,<mol>,"mol",<prob>

This command allows to replace atoms and molecules. The first two variations operate on atoms. The first variation replaces the atom at index <index> with atom name (or number) <at>. If the atom type does not already exist in the crystal, the temperature factor <dbw> has to be given as additional parameter. For this command variation, the atom no. <index> is replaced, no matter what its individual properties are and whether it fulfills the global property selection rules ==> 'property'

Alternatively the atom type <at1> can be replaced by <at2> within the complete crystal with a probability of <prob>. This works much (!) faster than replacing using DISCUS loops. The third parameter specifies if all "all" occurring atoms of type <at1> might be replaced or just

those on the given site <site>. For this replacement by probabilities, only those atoms are included that fulfill the global property selection rules ==> 'property'.

For both variations, if the atom is replaced by a void, the the property NORMAL is turned off, if a void is replaced by another atom type the property NORMAL is turned on.

The last two variations of the 'replace' command operate on molecules. The third variant replaces molecule <mo1> by <mo2>. The positions are exchanged relative to the origin of the molecule (atom 1) and both molecules MUST have the same length. The last variation works the same way, just all molecules of type <mtyp> will be replaced with molecule <mol> with the given probability of <prob>.

## A.42 rmc

### rmc

Enter "reverse Monte Carlo" section of discuss. This part of the program allows to "fit" a model atom configuration to measured diffuse scattering data. The difference between calculated scattering and observed scattering is minimized. Moves minimizing this difference are always accepted, other moved only with a probability of  $\exp(-\text{diff}^2)$ .

(Nield et al., Acta Cryst A51 (1995), 763-771 for more information)

### commands

Valid commands at this sub level are:

```
@      ! Executes a macro (see main help)
=      ! Algebra (see main help)
data   ! read experimental data for RMC fit
dese   ! Deselects the atomtypes allowed for RMC moves/switches (-> sele)
continue! Coninue a stopped macro (see main help level)
echo   ! Echos a string, just for interactive check
eval   ! Evaluates an expression (see main help)
exit   ! Terminates the rmc sub level, returns to the main DISCUS level.
help   ! Gives on-line help for 'rmc' (see main help)
mdes   ! Deselects molecules
msel   ! Selects molecules for RMC refinement
rese   ! reset RMC settings (i.e. reset plane counter)
run     ! start RMC fit
save   ! saves structure or intensity to given file
sele   ! selects the atomtypes allowed for RMC moves/switches (-> dese)
set    ! sets most RMC parameters
show   ! show current RMC settings
stop   ! Stops execution of a macro (see main help level)
system ! executes operating system command (see main help)
wait   ! Waits for user input (see main help)
```

### data

```
data {"x-ray"|"neutron"},{<name>|<value>},<file>,<wic>,
<e11>,<e12>,<e13>,<e21>,<e22>,<e23>,<e31>,<e32>,<e33>
```

This command reads the experimental data stored in <file>. The parameter "x-ray" / "neutron" selects what kind of scattering data are stored in the file. The second parameter is the wavelength which can be given as <name> (like cua1 ..) or as value (line 1.23). After the filename <file> the weighting scheme to be used can be specified. The parameter <wic> can take the following values:

```
"one"  : w(Int) = 1.0           "squa" : w(Int) = Int**2
"sqr"   : w(Int) = Sqrt(Int)    "inv"   : w(Int) = 1/Int
"log"   : w(Int) = LOG(Int)     "isq"   : w(Int) = 1/Sqrt(Int)
"lin"   : w(Int) = Int
```

All other values for <wic> are taken as the name of a file containing the weights. This file and the data file have to be the same type ! Finally the three corners of the input data plane are given in the following order:

```
<e11>,<e12>,<e13> : hkl's of the lower left corner
<e21>,<e22>,<e23> : hkl's of the lower right corner
<e31>,<e32>,<e33> : hkl's of the upper left corner
```

The format of the input file is set by the 'set data' command. For more information about the supported data file formats see help for the 'set data' command.

By repeating the data command you can read several planes with exp. data for the RMC fit. To start again use the 'rese' command.

## dese

**dese { "all" | <name> | <number> }, [ ... ]**

This command deselects atomtypes given either by <name> or <number> for the RMC moves, i.e. 'dese' those atomtypes which should not be moved or switched.

## msel

**msel { "all" | <number> } [ , <number> ... ]**

This command allows to select the molecule types to be used for the RMC refinement. By using 'msel' the atom selection becomes invalid. The parameter "all" will select ALL molecule types. Alternatively individual molecule types can be selected using the corresponding <number>. In order to use atoms again, use the 'sele' command.

## mdes

**mdes { "all" | <number> } [ , <number> ... ]**

This command deselects all or individual molecule types used in the RMC refinement.

## rese

### rese

This command resets the RMC parameters. In this version only the 'plane counter' is reset and the 'data' command will start reading plane 1 again.

**run****run**

Start the RMC fit. After the run is finished, the following values are stored in the res[i] variables:

```
res[1] : last value of CHI2
res[2] : number of tried moves
res[3] : number of accepted 'good' moves
res[4] : number of accepted 'bad' moves
res[5] : DELTA CHI2 average during the last run
res[6] : DELTA CHI2 standard deviation during the last run
res[7] : DELTA CHI2 maximum value during the last run
res[8] : elapsed time / cycle in seconds
```

**save****save <subcommand>**

This command is for saving the calculated intensities or the resulting structure. Valid subcommands are:

**lots****save "lots",<name>**

This command saves the current set of lot origins to the file named <name>. These lot origins can be used as default in subsequent refinement rather than creating a new set of random lots (-> 'set lots,..').

**stru****save "stru",<name>**

This command saves the current structure to the file named <name>. The format is identical to the unit cell or structure file. Each atom is written in format A4,3(2X,F11.6),5X,F6.4. The filename can be created by an format string and one or more numbers -> filenames.

**scat****save "scat",<plane>,<name>**

save "scat" saves the current computed intensities of plane <plane> in the following format: h,k,l,skal\*(a\*\*2+b\*\*2)+back (= intensity).

**sym****save "sym",<plane>,<sym>,<name>**

Does the same as 'save scat,..', but the user can save also the resulting intensities of symmetric equivalent planes by giving the number of the sym.-transformation (1=data corresponding to input data).

**sele**

**sele** { "all" | <name> | <number> }, [ ... ]

This command selects atomtypes given either by <name> or <number> for the RMC moves, i.e. 'sele' those atomtypes which should be moved or switched during the RMC cycles.

**set**

**set** "subcommand"

This command allows to set most of the RMC parameters. The following "subcommands" are valid:

**commands**

Valid subcommands are:

```
"aver"      : sets the portion of the crystal to be used for <F> calc.
"back"      : controls if background is calculated during RMC cycles
"const"     : sets possible constrains for scaling factor/background
"cycl"      : sets number of RMC cycles to be calculated
"data"      : sets data type of input/output diffuse scattering files
"dbw"       : ignore/use temperature factor
"disp"      : sets output interval
"log"       : controls if "rmc.log" will be created during calculation
"mdis"      : sets minimal allowed distances between atoms
"mode"      : sets RMC mode (relaxation/switch atoms)
"move"      : sets sigma for generated RMC shifts
"range"     : sets Q-range of data used for RMC refinement
"scal"      : controls if scaling factor is calculated during RMC cycles
"sigm"      : sets SIGMA for CHI2 calculation
"sym"       : controls if symmetric equivalent planes of input data should
               be used for RMC calculations.
```

**aver**

**set "aver",<per>**

This command sets the portion of the crystal, that should be used to calculate <F>. The variable <per> is given in %. The average structure factor <F> is calculated and subtracted from the overall structure factor calculated at the beginning of the RMC refinement. Starting from these values the contributions of the changes atoms during the RMC cycles are calculated and the structure factor adjusted. There is no check weather <F> changes significantly during the refinement.

To switch the use of <F> off, just set the value of <per> to zero. After setting the value of <per> the structure factor will be recalculated. This can also be used to recalculate the Fourier transform and the average from time to time during the RMC refinement if the average structure is changing.

**back**

**set "back",{"on"|"off"},[<value>],[<plane>]**

This command controls if a background parameter should be calculated and used for the RMC fits. If the calculation of the background parameter is switched off its value can be set to <value> otherwise the default is 0. The second optional parameter allows to specify the plane the current value of <value> is set for. If the parameter <plane> is omitted, the value of <value> is set for ALL experimental data planes.

**const**

**set "const",<p1>,<p2>**

This command allows to constrain the scaling factor and background parameter of the two input data planes number <p1> and <p2>. If <p1> and <p2> are equal, no constrain is set, and the planes have their individual scaling factor/background. If two planes should have the same values for scaling factor/background, the constrain is set by <p1> not equal <p2>. The value of <p1> is the plane which should take the values for scaling factor/background of plane <p2>. Not, that the value of <p1> has to be smaller than <p2>.

If for example scaling factor/background for plane 2 should be the same as for plane 1, use the command 'set const,2,1'. If there are more than two planes to constrain use the command several times. Plane 1 to 3 can be constrained by 'set const,2,1' and 'set const,3,2'. to unconstrain the planes again type 'set const 1,1', 'set const,2,2' and 'set const,3,3'.

**cycl**

**set "cycl",<icyc>**

This command sets the number of RMC moves to be done after the 'run' command. Note that the number given is the actual number of RMC moves to be tried and \*NOT\* the number of RMC cycles, which is normally defined as the number of moves necessary to visit every crystal site once on average.

**data**

**set "data",{ "nipl" | "pgm" }**

This commands sets the data type of the experimental data files used as input for RMC and the corresponding output files. There are currently two file formats supported:

**"nipl" : An ASCII file containing the Z-values in the following format:**

```

nx ny                ! # pixels in x- and y direction
xmin xmax ymin ymax  ! dimensions in x- and y-direction (ignored)
data                 ! ny rows of nx intensity values

```

**"pgm" : PGM file (pbmplus package). In the current version, only ASCII PGM files are supported. To convert the binary files into the ASCII PGM version, use the 'pnmnoraw' program, which is part of the 'pbmplus' package.**



**disp****set "disp",<id>**

This command sets the update interval of the output.

**dbw****set "dbw",{"on"|"off"} [,<plane>,...]**

This command controls whether the Debye-Waller factor is ignored or used when the scattering amplitude is calculated. If no parameter <plane> is given, the setting affects ALL read data planes.

**log****set "log",{"on"|"off"}**

This command controls if in the RMC main loop the screen output should also be logged to the file "rmc.log". This file contains only the last output and should give the possibility to watch the programs progress because cached output redirection might not update often enough.

**lots****lots "off"****lots {"box" | "ellipsoid"},<lx>,<ly>,<lz>,<n>,{"yes" | "no"},[<fname>]**

This command allows to use the 'lot' option during the RMC refinement. For details about the command and its parameters see the help file entry for 'four lots'.

The optional last parameter is a filename for a file containing a predefined set of lot origins. This file can be created using 'save lots,..' after a RMC refinement was carried out. If no filename <fname> is given, DISCUS creates a new random set of lots for the RMC fit.

The advantage of using lots is the combination of a smooth calculated diffraction pattern as average of the Fourier transform of small crystal volumes (lots). This forces RMC on a 'local' scale in real space without the problem of large errors and noisy patterns as for small crystal sizes.

It is important to subtract the average structure factor (-> set aver) when using this mode even if "periodic boundary conditions" are fulfilled, because the Fourier transform of the lots shape needs to be subtracted as well as <F>. This can be done using the command 'set aver ..'.

**mdis****set "mdis",{"all" | <name> | <number> },  
{ "all" | <name> | <number> },<dist>**

This command sets the minimal allowed distance between two atom types to the value of <dist> Å. A generated move will only be accepted if the moved atom is not closer than <dist> to the others. The atomtypes to be selected can be specified by either name or number or all atomtypes can be selected by typing "all".

## mode

```
set "mode",{ "shi[ft]" | "swc[hem]" | "swd[isp]" | "ext[ernal]" },
[ { "a[ll]" | "l[ocal]" | "sl[ocal]" | "si[te]" } ]
```

This command allows to select one of the three RMC operation modes:

```
"shift"      : The selected atoms are displaced by a random move. The
               max. size of a move can be specified via "set move .."
               command.
"swchem"     : Here two atoms are selected at random and their chemistry
               is switched (if the atoms are selected via "sel" command).
               If your structure consists of only ONE atom type this mode
               will just waste computer time :-)
"swdisp"     : Here the displacement from the average structure of two
               selected atoms of the SAME type are switched. The mode
               can only properly work, if there are initial displacements
               generated e.g. by the "therm" command.
"external"   : The modification of the structure for each RMC move is done
               using an user supplied external subroutine. Check the
               source file 'extrmc.f' for details how to supply such a
               subroutine.
```

The second optional parameter specifies if the pair of the atoms to be selected for the SWCHEM and SWDISP mode should be restricted in any way. The following settings are valid, the setting "all" is the default, which will be taken if no second parameter is entered.

```
"all"        : Any atom within the crystal can be chosen as second atom.
"local"      : Only atoms within +-1 unit cell from the first selected
               atom will be chosen.
"slocal"     : Same as "local" only the second atom must be on the same
               site within the unit cell as the first one.
"site"       : Atom on the same site in any unit cell are allowed as
               second atom.
```

## move

```
set "move",{ "all" | <name> | <number> },<sx>,<sy>,<sz>
```

Sets an user defined sigma for the generates moves. The created shifts are Gaussian distributed. The default is a sigma of 0.2 unit cell. Note that the values of <sx>, <sy> and <sz> are given in unit cell units to speed up the RMC runs. In case molecules are selected <number> specifies the molecule type, otherwise atoms are addressed by their <number> or <name>.

## range

```
set "range","all"
set "range",<qmin>,<qmax>
set "range",<hmin>,<kmin>,<lmin>,<hmax>,<kmax>,<lmax>
```

This command allows to set a range in Q which will be used for the RMC refinements. This can be useful if e.g. occupational disorder should only be refined based on the low Q part of the input data whereas the displacements should be refined using the complete diffuse scattering data set.

There are 3 ways to specify the Q range: "all" will use the complete input data set for the fit. The Q-range can be given as <qmin>, <qmax> ( in reciprocal A) or as hkl's (<hmin>,...,<lmax>).

### scal

**set "scal",{"on"|"off"},[<value>],[<plane>]**

This command controls if a scaling factor should be calculated and used for the RMC fits. If the calculation of the scaling factor is switched off it can be set to the value given by <value>. If no value is given, the default is 1. The second optional parameter allows to specify the plane the current value of <value> is set for. If the parameter <plane> is omitted, the value of <value> is set for ALL experimental data planes.

### sigm

**set "sigm",<sigma>**

This command sets the value for SIGMA. CHI2 is calculated by

$$\text{CHI2} = \text{SUM} (\text{Fobs}-\text{Fcalc})^2/\text{SIGMA}^2$$

The value of SIGMA is important for the calculation. A too large value will allow nearly every move and a too small value will drive the fit in the nearest local minimum. The default value is 0.01 If the value of SIGMA is set to zero, only moved improving the fit will be accepted.

For test and debug runs a setting of SIGMA to -9999. will result in the acceptance of ALL moves.

### sym

**set "sym",{"on"|"off"}**

This commands controls the use of symmetric equivalent planes to the read data-planes for the RMC calculations. If this flag is set to "off", only the experimental data are used for CHI2 calculation, if the flag is "on" the symmetry of the measured data is taken into account.

### show

**show [<subcommand>]**

Shows the current parameter settings of the RMC subsection. The following subcommands allow to select certain parameters of interest. The allowed subcommands are:

```
"all" : show all settings (same as calling 'show' without parameter)
"atom" : show selected atoms, minimal distances, ...
"data" : show information about experimental data
"mode" : show general RMC calculation settings
"rval" : show R-values for current refinement
"sym" : show information about using symmetry information of read planes
```

## A.43 save

**save** <name>

saves the current structure to file <name>

The format is identical to the unit cell or structure file. Each atom is written in format A4,3(2X,F11.6),5X,F6.4

If the 'save' command is used with one parameter, the parameter is interpreted as output file name. See the entry ==> 'filenames' under the command language entry in the main help section on information about valid file names and file name substitution. The save menu is executed using the current default values and the standard DISCUS prompt returns.

If the 'save' command is used without parameter, the save menu is called. At this menu you can define the output format of the structure file, its name, which information and which atoms to include.

### commands

Valid commands at this level are

```
@      ! Execute a macro file (see main help)
=      ! assigns the value to a variable (see main help)
asym   ! Shows asymmetric unit
chem   ! Shows the atoms present in the crystal
continue! Coninue a stopped macro (see main help level)
des    ! deselects atoms
echo   ! echo a string (see main help)
eval   ! Evaluates an expression for interactive check (see main help)
exit   ! terminates 'save'
file   ! Defines the file name to which the structure is written
format ! Defines the format of the structure file
help   ! gives on line help for saving the structure (see main help)
omit   ! Defines which information to omit from the file
run    ! starts the saving of the structure
sel    ! selects atoms to be included in the structure file
show   ! shows the current parameters
stop   ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
wait   ! Waits for user input (see main help)
write  ! Defines which information to write to the file
```

### asym

#### asym

Shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the variable m[<index>]. If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

**chem****chem**

Displays the type of atoms present in the crystal. For each type of atom, its scattering curve number, its name and its temperature factor are listed. Warning, even, if all atoms of a particular type have been deleted, its scattering type will remain in the list. This list could therefore include more types of atoms than are actually present in the crystal.

**des**

**des** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'sel'. Possible values for the parameter are mutually exclusively:

```
"all"      all atoms of the crystal are deselected.
<name>     all the atoms called <name> of the crystal are deselected.
            This includes symmetrically not equivalent atoms.
<number>   all atoms of the crystal that are of scattering type <number>
            are deselected.
```

More than one atom may be deselected at once.

**outfile**

**outfile** <filename>

Sets the name of the output file to which the structure is written.

**format**

**format** {"keyword" | "nokeyword" }

The keyword controlled format uses keywords analogous to the commands of DISCUS to describe the information in the structure file. See the help on "data" at the main help level for a complete listing of all keywords. A flexible number of keywords can be included at the header of the structure file ==> 'write' and 'omit'.

The non-keyword controlled format writes a simplified file that contains only the following information:

```
Line 1:  Title up to 80 characters
line 2:  Space group symbol [,origin choice number]
line 3:  Lattice constants a,b,c,alpha,beta,gamma
```

and for each atom a line with:

```
Name(4 characters) x y z isotropic B
```

**incl**

```
incl { <start>,<end> | "all" }
```

The unit cell transformation includes all atoms numbered <start> to <end> inclusively. All other atoms are ignored. If, instead of explicit numbers, the parameter "all" is given, the unit cell transformation will include all atoms of the crystal. This holds even, if at a later time you include further atoms in the crystal. Thus, you can define a setup for the unit cell transformations early in a lengthy macro, then modify the crystal and just run the unit cell transformation later on. In addition you can define the atoms that are affected by the unit cell transformation operation with the ==>'sele' and 'dese' commands.

**omit**

```
omit { "all" | "generator" | "ncell" | "symmetry" | "molecule" |  
"object" | "scat" | "adp" }
```

The keyword controlled format includes the mandatory lines : title spcgrp cell atoms  
All other keywords are optional.

"all" All optional keywords are omitted from the structure file.

"scat" The list of all atom names is omitted from the structure file, while the status of the other keyword is not changed.

"adp" The list of all Atomic displacement parameters is omitted from the structure file, while the status of the other keyword is not changed.

"generator" The additional generators are omitted from the structure file, while the status of the other keyword is not changed.

"ncell" The number of unit cells and the number of atoms in each unit cell are omitted from the structure file, while the status of the other keyword is not changed.

"symmetry" The additional symmetry operators are omitted from the structure file, while the status of the other keyword is not changed.

"molecule" Although the crystal contains molecules their type and content is not written to the end of the structure file. WARNING: All information about molecules is lost!

"object" Although the crystal contains objects their type and content is not written to the end of the structure file. WARNING: All information about objects is lost!

**run****run**

Starts the transformation operation.

**sel**

```
sel { "all" | <name> | <number> } [, { <name> | <number> } ...]
```

This command serves to select those atoms that will be modified by the unit cell transformation. Defines which atoms are included in unit cell transformations. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all atoms of the crystal are included.
```

This includes the "voids" in the structure, which are stored as scattering curve number zero.

```
<name>    all the atoms called <name> of the crystal are included.
           This includes symmetrically not equivalent atoms.
<number>  all atoms of the crystal that are of scattering type <number>
           are included.
```

More than one atom may be selected at once.

The selection made stay valid until explicitly deselected!

## show

### show

Shows the current parameters of the transformation operation.

## write

```
write { "all" | "generator" | "ncell" | "symmetry" | "molecule" |
        "object" | "domain" | "scat" | "adp" }
```

The keyword controlled format includes the mandatory lines : title spcgrp cell atoms  
All other keywords are optional.

"all" All optional keywords are written to the structure file.

"scat" The list of all atom names is written to the structure file.

"adp" The list of all atomic displacement parameters is written to the structure file.

"generator" If additional generators were read from the input file they will be written to the structure file as well.

"ncell" The number of unit cells and the number of atoms in each unit cell are written to the structure file. The number of unit cells corresponds to the parameter used for the 'read/cell' command. These numbers are needed for a fast association between atom number and its unit cell inside a large crystal. Warning!!!! If you purged "voids" from the crystal, the number of atoms per unit cell is no longer constant throughout the crystal and no fixed association exists between atom number and the corresponding unit cell. Use of the purge command is discouraged !!!!!

"symmetry" If additional symmetry operators were read from the input file they will be written to the structure file as well.

"molecule" If the crystal contains molecules their type and content is written to the end of the structure file.

"object" If the crystal contains objects their type, content, and density is written to the end of the structure file.

"domain" If the crystal contains domains their type, character, and content is written to the end of the structure file.

## A.44 seed

```
seed [ <value> ]
```

Reinitializes the pseudo random generator. The seed passed to the random generator is -abs(nint(value)). If the <value> is omitted, the random generator will be passed the number of hundredth of seconds passed since midnight, essentially initializing the sequence at a unknown fairly random point.

## A.45 shear

### shear

This command enters the deformation menu. The commands in this subsection allow you to apply a deformation tensor to an object or to a molecule.

You can define the deformation tensor by providing: - the matrix - a deformation plane and the deformation vector parallel to this plane. - the eigenvectors and eigenvalues of the deformation tensor

Further help is available on:

### commands

```
calc      ! Calculate the deformation for a single vector
des       ! Deselect the atom types
eigen     ! Define the deformation through it's eigenvectors
hkl       ! Define the deformation plane
incl      ! Include a range of atoms
matrix    ! Define the deformation tensor directly
mincl     ! Include a range of molecules
mdes      ! Deselect the molecule types
msel      ! Select molecule types
mode      ! Set the input mode (matrix, plane, eigenvectors)
oincl     ! Include a range of objects
origin    ! Define the origin for the deformation
osel      ! Select the object types
odes      ! Deselect the object types
run       ! Perform the actual deformation
sel       ! Select the atom types
show      ! Show the current settings
uvw       ! Define the deformation plane
vector    ! Define the deformation vector
```

### calc

**calc** <x>,<y>,<z> {,<flag>}

Calculates the effect of the current shear on vector <x>,<y>,<z>. The flag is allowed to be either "d" or "r", specifying a direct or reciprocal vector. The result is displayed on screen and stored in the result variable "res[]". 3 values are stored, <x'>,<y'>,and <z'>.

### des

**des** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'sel'. Possible values for the parameter are mutually exclusively:



```

"all"      all atoms of the crystal are deselected.
<name>     all the atoms called <name> of the crystal are deselected.
           This includes symmetrically not equivalent atoms.
<number>   all atoms of the crystal that are of scattering type <number>
           are deselected.

```

More than one atom may be deselected at once.

## eigen

**eigen** <nr>, <x>, <y>, <z> {,<value>}

Define the direction of the eigenvector number <nr>. <x>, <y>, <z> are the coordinates of the eigenvector in direct space.

If you give the last parameter, this is interpreted as the corresponding eigenvalue. If the last parameter is omitted, DISCUS will calculate the length of the eigenvector and take this as the eigenvalue.

In both cases, the eigenvector is normalized to unit length.

The ==> 'show' command displays the Eigenvectors, the determinant of the eigenvector matrix, and the angles between the three eigenvectors. This allows you to verify that all three vectors are normal to each other.

## hkl

**hkl** <h>, <k>, <l> [, <length>]

Defines the deformation by providing a plane ==> 'hkl', 'uvw' and a deformation vector ==> 'vector'.

'hkl' is the normal to the plane in reciprocal space coordinates.

The object experiences a shear parallel to the plane. The magnitude of the shift is defined by the deformation vector, that shifts the point at the end of 'uvw' from the ==> 'origin' by its value ==> 'vector'. If the optional last parameter is given, the length of the normal ==> 'uvw' to the plane is normalized to this length, otherwise the normal is taken as is. This parameter helps you e.g. in a monoclinic space, to apply a shift parallel to the (001) plane by specifying the ==> 'hkl' 0,0,1 and then giving a distance from this plane at which the shear vector ==> 'vector' is to act, e.g. the d-spacing of the (001) planes. All points are shifted by an amount parallel to the deformation vector that is proportional to their distance from the plane. Points at a distance equal to the length of 'uvw' are shifted by the vector, # those twice as far away are shifted twice as much. Points on opposite sides of the plane are shifted in opposite directions.

## incl

**incl** {<start>, <end> | "all" | "envi" }

The shear operation includes all atoms numbered <start> to <end> inclusively. All other atoms are ignored. If, instead of explicit numbers, the parameter "all" is given, the shear operation will include all atoms of the crystal. This holds even, if at a later time you include further atoms in the crystal. Thus, you can define a setup for the shear operations early in a

lengthy macro, then modify the crystal and just run the shear operation later on. If the parameter "envi" is specified, the shear operation will include all atoms in the current environment that was located by the last ==> 'find env' command. In addition you can define the atoms that are affected by the shear operation with the ==>'sele' and 'dese' commands.

### matrix

**matrix** <11>,<12>,<13>,<21>,<22>,<23>,<31>,<32>,<33>

This command defines explicitly the deformation tensor. All nine elements must be given.

### mincl

**mincl** {<start>,<end> | "all" }

The shear operation includes all molecules numbered <start> to <end> inclusively. All other molecules are ignored. If, instead of explicit numbers, the parameter "all" is given, the shear operation will include all molecules of the crystal. This holds even, if at a later time you include further molecules in the crystal. Thus, you can define a setup for the shear operations early in a lengthy macro, then modify the crystal and just run the shear operation later on. In addition you can define the molecules that are affected by the shear operation with the ==>'msele' and 'mdese' commands.

### mdes

**mdes** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'msel'. Possible values for the parameter are mutually exclusively:

```
"all"      all molecules of the crystal are deselected.
<number>  all molecules of the crystal that are of type <number>
           are deselected.
```

More than one molecule type may be deselected at once.

### msel

**msel** { "all" | <number> } [, <number> ...]

This command serves to select those molecules that will be modified by the symmetry operation.

Defines which molecules are included in symmetry operation. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all molecules of the crystal are included.
           This includes the empty molecules in the structure, which are
           stored as scattering curve number zero.
<number>  all molecules of the crystal that are of type <number>
           are included.
```

More than one atom may be selected at once.

The selection of molecules by 'msel', objects by 'osel', and atoms by 'sel' are mutually exclusive. The symmetry operation uses either atoms, objects or molecules.

**mode**

**mode** {"eigen"|"matrix"|"plane"|"rmatrix"}

You can tell DISCUS which input you intend to use for the definition of the deformation tensor. The commands ==> 'uvw', 'hkl', 'vector', 'matrix', 'rmatrix', and 'eigen' will automatically set this flag as well. The last of these commands type before the final ==> 'run' determines the actual input mode that is assumed.

**odes**

**odes** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'osel'. Possible values for the parameter are mutually exclusively:

```
"all"      all objects of the crystal are deselected.
<number>  all objects of the crystal that are of type <number>
           are deselected.
```

More than one objects type may be deselected at once.

**oincl**

**oincl** {<start>,<end> | "all" }

The shear operation includes all objects numbered <start> to <end> inclusively. All other objects are ignored. If, instead of explicit numbers, the parameter "all" is given, the shear operation will include all objects of the crystal. This holds even, if at a later time you include further objects in the crystal. Thus, you can define a setup for the shear operations early in a lengthy macro, then modify the crystal and just run the shear operation later on. In addition you can define the objects that are affected by the shear operation with the ==>'osele' and 'odese' commands.

**origin**

**origin** <x>,<y>,<z> [,{"crystal"|"molecule"}]

This commands sets the origin to which the shear operation refers to.

If you selected atoms by ==> 'sel', the last optional parameter will always default to an origin with respect to the crystal base system. The specified origin is subtracted from each atom position prior to the shear operation.

If you selected molecules ==> 'msel' or objects ==> 'osel', the last optional parameter will be interpreted. If it is set to "molecule", the origin is considered to be at vector <x>,<y>,<z> relative to the origin of each individual molecule or object. All distances within the molecule or object are sheared accordingly. If it is set to "crystal", the origin is considered to be at vector <x>,<y>,<z> relative to the crystal origin. The shear operation is applied to the origin of each molecule or object. The molecules or objects are now considered to be rigid units. Thus the shear operation will shift all atoms or objects with the shear that is applied at their respective origin but will not apply a shear to the internal distances.

## osel

**osel** { "all" | <number> } [, <number> ...]

This command serves to select those objects that will be modified by the symmetry operation. Defines which objects are included in symmetry operation. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all objects of the crystal are included.  
           This includes the empty objects in the structure, which are  
           stored as scattering curve number zero.  
<number>  all objects of the crystal that are of type <number>  
           are included.
```

More than one object may be selected at once.

The selection of objects by 'osel', 'msel' and atoms by 'sel' are mutually exclusive. The symmetry operation uses either atoms or molecules, or objects.

## run

**run**

Performs the actual deformation operation on the selected atom, molecule or object types.

## sel

**sel** { "all" | <name> | <number> } [, { <name> | <number> } ...]

This command executes serves to select those atoms that will be modified by the symmetry operation.

Defines which atoms are included in symmetry operation. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all atoms of the crystal are included.
```

This includes the "voids" in the structure, which are stored as scattering curve number zero.

```
<name>     all the atoms called <name> of the crystal are included.  
           This includes symmetrically not equivalent atoms.  
<number>  all atoms of the crystal that are of scattering type <number>  
           are included.
```

More than one atom may be selected at once.

The selection made stay valid until explicitly deselected!

## show

**show**

Shows the current settings.

**uvw****uvw** <u>,<v>,<w> [,<length>]

Defines the deformation by providing a plane ==> 'hkl', 'uvw' and a deformation vector ==> 'vector'.

'uvw' is the normal to the plane in direct space coordinates.

The object experiences a shear parallel to the plane. The magnitude of the shift is defined by the deformation vector, that shifts the point at the end of 'uvw' from the ==> 'origin' by its value ==> 'vector'. If the optional last parameter is given, the length of the normal ==> 'uvw' to the plane is normalized to this length, otherwise the normal is taken as is. This parameter helps you e.g. in a monoclinic space, to apply a shift parallel to the (001) plane by specifying the ==> 'hkl' 0,0,1 and then giving a distance from this plane at which the shear vector ==> 'vector' is to act, e.g. the d-spacing of the (001) planes. All points are shifted by an amount parallel to the deformation vector that is proportional to their distance from the plane. Points at a distance equal to the length of 'uvw' are shifted by the vector, # those twice as far away are shifted twice as much. Points on opposite sides of the plane are shifted in opposite directions.

**vector****vector** <x>,<y>,<z>

Defines the deformation by providing a plane ==> 'hkl', 'uvw' and a deformation vector ==> 'vector'.

The object experiences a shear parallel to the plane. The magnitude of the shift is defined by the deformation vector, that shifts the point at the end of 'uvw' from the ==> 'origin' by its value ==> 'vector'. All points are shifted by an amount parallel to the deformation vector that is proportional to their distance from the plane. Points at a distance equal to the length of 'uvw' are shifted by the vector, # those twice as far away are shifted twice as much. Points on opposite sides of the plane are shifted in opposite directions.

**A.46 show****show** <subcommand>,..

This command allows to obtain various information about the model crystal, DISCUS settings etc.

Valid subcommands are :

**asym****show "asym"**

shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the variable m[<index>]. If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was

read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

### atom

**show "atom",<number> [,<end>]**

**show "atom","all"**

**show "atom","last"**

The type, position and isotropic B of atom number <number> are shown. If the optional second parameter <end> is given, all atoms in the range <number> to <end> are shown. If the second parameter is "all", all atoms in the crystal are shown. WARNING, this could last a while :-). If the second parameter is "last", only the last atom of the crystal is shown. This is identical to setting the second parameter to "n[1]", which contains the number of atoms in the crystal.

If <number> is higher than the number of atoms in the crystal, nothing is shown and no error message is displayed. If <end> is higher than the number of atoms in the crystal, <end> is reset to the last atom in the crystal.

### bval

**show "bval",<a1>,<a2>**

This command shows the bond valence parameters  $r_0$  and  $b$  for the specified atom pair <a1> and <a2>. The parameters are taken from a table prepared by I.D. Brown, McMaster University, ON, Canada.

### chem

**show "chem"**

Displays the type of atoms present in the crystal. For each type of atom, its scattering curve number, its name and its temperature factor are listed. Warning, even, if all atoms of a particular type have been deleted, its scattering type will remain in the list. This list could therefore include more types of atoms than are actually present in the crystal. A short information about the content of the asymmetric unit is printed.

### cdim

**show "cdim"**

Prints the current crystal dimensions. The smallest and largest coordinates of any atom along x,y and z are printed.

### config

**show "config"**

This command lists the current configuration of DISCUS such as the maximum number of atoms.

**connectivity**

**show "connectivity", <atom\_no.>, <def.no.>**

Show the list of neighbors around atom no. <atom\_no.>, which have been found using definition number <def.no.> for the scattering type of atom no. <atom\_no.> The list of atom numbers is copied into the result variable "res".

**envi**

**show "environment"**

All atoms in the current environment ==> 'find env' are listed.

**error**

**show "error"**

The error status of DISCUS is displayed.

**menv**

**show "menviroment" [, "full"]**

All molecules in the current environment ==> 'find env' are listed. If the optional "full" parameter is given, all atoms within each molecule are printed.

**metric**

**show metric**

Displays the current unit cell dimensions, the unit cell volume, the metric tensor and the corresponding reciprocal space values. This command is for display only. You may change the direct space unit cell dimensions with the assignment `lat[<i>]=<value>`. <i>=1 through 3 are a,b,c and <i>=4 through 6 are alpha, beta, gamma.

**molecule**

**show "molecule",{ "all" | "last" }**

**show "molecule", <number> [, <end> ]**

For each molecule in the range <number> to <end> the type, number of the atom in the crystal, position and isotropic B of all atoms in the molecule are shown.

If the second parameter is "all", all molecules in the crystal are shown. WARNING, this could last a while :) If the second parameter is "last", only the last molecules of the crystal is shown.

If <number> is higher than the number of molecules in the crystal, nothing is shown and no error message is displayed. If <end> is higher than the number of molecules in the crystal, <end> is reset to the last atom in the crystal.

**object****show "object",{"all" | "last"}****show "object",<number> [,<end>]**

For each object in the range <number> to <end> the type, number of the atom in the crystal, position and isotropic B of all atoms in the object are shown.

If the second parameter is "all", all objects in the crystal are shown. WARNING, this could last a while :-)

If the second parameter is "last", only the last objects of the crystal is shown.

If <number> is higher than the number of objects in the crystal, nothing is shown and no error message is displayed. If <end> is higher than the number of objects in the crystal, <end> is reset to the last atom in the crystal.

**result****show "result"**

The result array is shown on screen. The number of values shown depends on the last calculation performed.

Further information about the crystal, its composition etc can be obtained by evaluating the corresponding ==> variable.

**scat****show "scat", {<name> | <number> } [, {<name> | <number> }...]**

Shows the scattering curve of the selected atom types, which may be given as the scattering number of as the element name.

**symmetry****show "symmetry" [,{"full" | "symbol" | "xyz" | "matrix" }]**

Shows all symmetry operations for the current space group.

**"full"**

The default setting, produces a full listing. This includes: the symmetry matrix as (3x4) matrix the symbol as printed in the International Tables the resulting atom positions xyz, as printed in the International Tables

For each symmetry operation DISCUS lists the number in square brackets, and for centered space groups the number within each block that corresponds to one centering vector in round brackets. With the setting ==> set generator,symmetry the sequence corresponds to that listed in the Tables.

**"symbol"**

This produces a short list that consists of the symbol as used in the International Tables.

**"xyz"**



This produces a short list that consists of the symbolic atom positions xyz, as used in the International Tables.

`"matrix"`

This produces a list that consists of the (3x4) symmetry matrices.

## A.47 stack

### stack

Switches to the stacking fault submenu of DISCUS. At this sub menu you create a crystal containing stacking faults. Several types of layers can be stacked on top of each other. At present these layers are read from file. In future releases you will be able to call a macro that creates the layer. A layer could be a 2-dimensional layer of atoms or unit cells of any thickness or a 1-dimensional row of atoms or unit cells. The translation vectors from one layer type to the next can be defined and need not be exact values but can be Gaussian distributed. A correlation matrix defines the probabilities of layer sequences. These probabilities only include first neighbor interactions.

Alternatively to the stacking faults ruled by the correlations, you can include random stacking faults. These stacking faults choose at random the next layer type and the translation vector.

You can create the crystal following two alternate choices:

- 1) The origin of the current layer is the sum of the previous origin and the translation vector
- 2) The above origin is taken modulo the modulus vectors.

By using the first option, the crystal continuously grows in one direction as given by the translation vector(s). By using the second option the origins are restricted to a finite range around the average translation vector, which results in a zig-zag shaped crystal.

The menu includes a new approach to calculate the Fourier transform that saves considerable computational time.

Further help topics are

### commands

Valid commands at this level are

```
@      ! Execute a macro file (see main help)
=      ! assigns the value to a variable (see main help)
aver   ! Sets the intended average translation
ccol   ! Reads a single column of the correlation matrix
cele   ! Reads a single element of the correlation matrix
continue! Coninue a stopped macro (see main help level)
crea   ! Creates the list of layer origins
crow   ! Reads a single row of the correlation matrix
dist   ! Defines the distribution of layer types
echo   ! echo a string (see main help)
eval   ! Evaluates an expression for interactive check (see main help)
exit   ! terminates 'stack' (see main help)
four   ! calculates the Fourier transform in quick mode
help   ! gives on line help for symmetry operations (see main help)
layer  ! Reads the name of a file that contains a layer
modulus ! Reads the modulus vector of the translation
```

```

number ! Reads the number of layers to be created
random ! Sets values for random stacking faults
rese   ! Resets the number of layers to zero
rotate ! Set values for rotational disorder of the layers
run     ! Decorates the list of origins with atoms to create the crystal
set     ! Sets parameters
show    ! Shows the current settings
sigma  ! Reads the sigmas for the translation vectors
stop    ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
trans  ! Reads the translation vectors
wait   ! Waits for user input (see main help)

```

**aver**

**aver** <u>,<v>,<w>

Sets the intended average translation. If the modulus is switched on ==> 'set modulus', and the mode of the translation ==> 'set trans' is set to "fixed", this vector is used.

**ccol**

**ccol** <nr>,<c1nr>,<c2nr>,...,<cmdtypenr>

Read a single column <nr> of the correlation matrix. Following the index <nr>, as many parameters as there are layer types must follow. The sum of any column need not be equal to one.

**cele**

**cele** <i>,<j>,<value>

Reads the value <value> that is stored in element <i>,<j> of the correlation matrix.

**crea**

**crea**

Creates the list of layer origins. No changes are done to the actual structure, this is performed by the command ==> 'run'.

**crow**

**crow** <nr>,<cnr>1,<cnr>2,...,<cnrmdtype>

Read a single row <nr> of the correlation matrix. Following the index <nr>, as many parameters as there are layer types must follow. The sum of any row need not be equal to one, as long as the sum of all rows is identical.

**dist**

**dist** { "matrix" | "file", <inputfile> | "list", <inputfile> }

This commands allows you to define how the sequence of layer types is determined. The parameter "matrix" forces DISCUS to determine the type of each layer from the correlation matrix ==> 'crow', 'ccol'. Only nearest neighbor correlations are realized.

The parameter "file" forces DISCUS to read a standard structure file from <inputfile>. The atom types found in this file are interpreted as to correspond to the respective layer type. Make sure this is a one-dimensional structure and that you have written this file using the ==> 'save' options to write the explicit list of scattering types and atomic displacement parameters. You can use the MonteCarlo section of DISCUS ==> 'mc' to create the list of layer types, which allows you to include nearest as well as next nearest etc neighbor correlations. Presently 'stack' ignores the atom positions found in the input file. The layer origins are calculated using the translations ==> 'tran'.

The parameter "list" acts very similarly to "file". The only difference is that the atom positions are interpreted as layer positions as well.

DISCUS does not check whether the atoms in <inputfile> form a 1-dimensional list of atoms, or whether this 1-dimensional line is not parallel to the layers! This can actually be used to distribute equally sized and oriented domains throughout the crystal. In combination with the stacking fault fourier option, very fast Fourier transformations of molecular crystals can be calculated.

**four****four**

A crystal with stacking faults can be created in the following step wise manner:

1. Create the distribution of layer origins.
2. Convolute each origin with the corresponding layer.

At this point, the convolution theorem saves a lot of computation time since the Fourier transform of a convolution is equal to the product of the individual Fourier transforms. The Fourier transform approach realized in the stacking fault menu operates by the following steps:

0. The current parameters of the ==> 'fourier' menu are used.
1. For each layer type calculate the Fourier transform of the list of origins.  
Calculate the Fourier transform of the actual atoms in the layer.  
Multiply the resulting complex amplitudes and store the results in temporary files.
2. Add the complex amplitudes from all layer types and store the result in internal memory.

The user can then write any desired output value using the ==>'output' sub menu.

**Note:**

In order to calculate the Fourier transform of each layer type the atoms in the corresponding file are read into the internal structure array. A previous structure is lost! If necessary save this structure to file by the ==> 'save' command.

**Note:**

DISCUS does not compare the space group or lattice constants in the different layer files. You must make sure that these are identical.

## layer

**layer** <filename>

Defines a new layer type. The file <filename> must be a standard DISCUS input file that contains a list of atoms. No restrictions apply to the position and types of atoms in the layer. You have to create the layer beforehand using the tools of DISCUS.

## modulus

**modulus** <x1>,<y1>,<z1>,<x2>,<y2>,<z2>

Sets the two moduli vectors that are used to restrict the origins to a finite volume around the average translation vector. The plane defined by these two vectors need not to be normal to the average translation.

```
==> 'set mod'
==> 'set tran'
==> 'aver'
```

You can create the crystal following two alternate choices:

- 1) The origin of the current layer is the sum of the previous origin and the translation vector
- 2) The above origin is taken modulo the modulus vector.

By using the first option, the crystal continuously grows in one direction as given by the translation vector(s). By using the second option one or two coordinates can be constrained to a finite range, which results in a zig-zag shaped crystal. If any of the parameters is not equal to zero, the corresponding coordinate of the origin is taken modulo this parameter.

DISCUS does NOT check whether the moduli vectors are translation vectors of the current space group!

## number

**number** <n>

Reads the number of layers to be created.

## random

**random prob**,<value>

**random offset**,< u>,< v>,< w>

**random sigma** ,<su>,<sv>,<sw>

This command sets values for the random stacking faults. The program selects the next layer type by weighted random choice. The weights are chosen according to the overall probabilities for each layer type. The next layer is then shifted by the corresponding translation vector

between the resulting neighbors. In addition a random fraction of the two moduli vectors is added, which results in a random distribution of the next layer origin.

**random prob,<value>**

Sets the probability for random stacking faults. <value> must be between zero and one. Set the probability to zero to exclude random stacking faults.

**random offset,< u>,< v>,< w>**

If specified this vector is added to the translation vector. You can use this offset to force an increased layer separation for random faults compared to regular stacking faults.

**random sigma ,<su>,<sv>,<sw>**

Specifies the sigma's for a Gaussian distribution applied to the components of the offset.

**rese**

**rese**

Sets the number of layers and layer types back to zero.

**rotate**

**rotate axis, "normal",<u>,<v>,<w> [ "d" | "r" ]**

**rotate axis, "mod1", <u>,<v>,<w> [ "d" | "r" ]**

**rotate axis, "mod2", <u>,<v>,<w> [ "d" | "r" ]**

**rotate mode, [ "all" | "fault" ]**

**rotate sigma, "normal",<snormal>**

**rotate sigma, "mod1", <smod1>**

**rotate sigma, "mod2", <smod2>**

**rotate status, [ "on" | "off" ]**

This command sets values for rotational disorder of the stacking faults. The layers are rotated around three axes, which are defined by the 'axis' command. The angle of rotation is Gaussian distributed with mean value zero and sigma defined by the 'sigma' command.

**rotate axis, "normal",<u>,<v>,<w> [ "d" | "r" ]**

**rotate axis, "mod1", <u>,<v>,<w> [ "d" | "r" ]**

**rotate axis, "mod2", <u>,<v>,<w> [ "d" | "r" ]**

The parameter "axis" defines the three axes around which you can rotate the layers. The axis can be given in real space coordinates "d" or reciprocal space coordinates "r". Default is real space. DISCUS does not make any assumptions about the three axes. The layer is rotated first around the "normal" axis, then around "mod1" and finally around "mod2". It is suggested to define "normal" as the axis perpendicular to the layers and "mod1" and "mod2" within the plane of the layers.

Attention!! If you tilt the layer around "mod1" or "mod2" the outer sections of the layer might overlap with the previous layer!

**rotate mode, [ "all" | "fault" ]**

The rotational disorder can affect all layers individually, regardless whether the previous layer was of the same type as the current layer or not. Alternatively, a new rotation angle is computed

only at a stacking fault, i.e. if the current layer type is different than the previous layer type. All following layers of identical type up to the next stacking fault are rotated by the same angle.

**rotate sigma, "normal", <snormal>**

**rotate sigma, "mod1", <smod1>**

**rotate sigma, "mod2", <smod2>**

The rotation is Gaussian distributed with individual sigma for each axis. The mean value of the distribution is zero. Each new angle is calculated individually from a Gaussian distribution irrespective of the previous value. This forces the average layer orientations to remain constant throughout the crystal. A sigma of zero fixes the rotations around the respective axis to zero.

**rotate status, [ "on" | "off" ]**

The rotational disorder is only effective if its status is switched "on".

## run

### run

Decorates the origins with the corresponding atom distribution. Any previously existing structure is lost.

## set

**set "aver", <value>**

**set "modulus", { "off" | "on" }**

**set "trans", { "calculate" | "fixed" }**

**set "aver", <value>**

If the value of <value> is not equal to zero, the average <F> is subtracted from the actual structure factor F. Otherwise the average <F> is calculated using the whole crystal and subtracted from the actual F.

```
==> "four"
```

```
==> "discus four set aver"
```

**set "modulus", { "off" | "on" }**

If the modulus is set "off", each origin is calculated by adding the corresponding translation vector to the previous origin. This way the crystal grows along the average translation vector. The resulting average translation vector is the sum of all translation vectors weighted by the corresponding probability as stored in the correlation matrix. If the modulus is set "on", each origin is calculated as above. The origin is then expressed as multiples of the two moduli vectors and the average translation. If any of the two multipliers corresponding to the two moduli vectors is larger than |1|, the moduli vector is subtracted from the origin. This has the effect that all origins are no further than the moduli vectors from the average translation. Even though all translation vectors might point in one direction, you can restrict the crystal to any arbitrary direction by defining the average vector ==> 'set trans', ==> 'aver'.

**set "trans", { "calculate" | "fixed" }**

You have two choices to restrict the crystal growth once the modulus is set on. The average translation can be fixed to the value given by ==> 'aver' by setting the parameter of the 'set

'trans' command to "fixed". Alternatively the average translation is computed using the array of the translation vectors and the correlation matrix.

## show

### show

Shows the current settings.

## sigma

**sigma** <i>,<j>,<sx>,<sy>,<sz>

Read the sigmas of the translation vector between layer type <i> and <j>. The current origin of any layer is calculated by adding the translation vector ==> 'trans' between layer types <i> and <j> to the previous origin. If any sigma is non-zero, a Gaussian distributed value with mean value of zero is added to the corresponding vector component.

## trans

**trans** <i>,<j>,<tx>,<ty>,<tz>

Reads the components of the translation vector between layers <i> and <j>. The current origin of any layer is calculated by adding the translation vector between layer types <i> and <j> to the previous origin.

## A.48 surface

Menu to define crystal surfaces and related settings Further help is available on:

### commands

```
boundary    ! Cut atoms that are outside a boundary
set         ! Make settings related to surfaces
show        ! Show the current surface related settings
```

## boundary

**boundary** "hkl",h,k,l [,distance] [,{"inside"}|{"outside"}]

**boundary** "sphere",<radius> [,{"inside"}|{"outside"}]

**boundary** "cylinder",<radius>,<half-height> [,{"inside"}|{"outside"}]

This commands removes all atoms that are outside a boundary surrounding the crystal. "hkl" The surface is defined as a plane through the Miller indices <h>,<k>,<l>. The optional <distance> determines the distance between 0,0,0 and the boundary surface. Without this parameter, the distance is automatically calculated from the user supplied <h>,<k>,<l> as 1./dstar(h,k,l). The optional last parameter determines which atoms remain. If the parameter is "inside" or omitted, all atoms remain that are on the same side of the plane as is the origin. If the parameter is "outside", those atoms remain that are on the other side of the plane as the

origin. "sphere" The boundary is defined as a sphere at the origin of direct space. The size of the sphere is defined by its radius in Angstroem. The optional last parameter determines which atoms remain. If the parameter is "inside" or omitted, all atoms remain that are inside the sphere. If the parameter is "outside", those atoms remain that are on the outside of the sphere. "cylinder" The boundary is defined as a cylinder of radius <radius> and half-height <height>, both measures are in Angstroem. The cylinder axis is always along the c-axis. The top and bottom planes are parallel to the (001) plane. If the a- and b-axes are not at 90 degrees to the c-axis, the top will be tilted accordingly. If necessary, transform into an orthogonal cell prior to the application of the boundary command. The optional last parameter determines which atoms remain. If the parameter is "inside" or omitted, all atoms remain that are inside the sphere. If the parameter is "outside", those atoms remain that are on the outside of the sphere.

## set

Make various setting related to surfaces

**set distance**, {"external" | "internal" | "all"}, <atom\_list>, <distance>

**set distance**, {"external" | "internal" | "all"}, {"default" | "off"}

Defines the distances between those atoms in <atom\_list> and the specified surface. All atoms of the selected type that are closer than <distance> to the specified surface are marked as "close". The respective property can be used everywhere an atom type is selected. ==> 'property' menu and the 'sel' / 'des' commands in various menus.

"external" The surface created by the ==> 'boundary' command, see the ==> 'surface' menu and the main menu 'boundary' command.

"internal" The surface is an internal surface between atoms that are inserted through a ==> 'domain' and the host structure.

"all" The distances apply to both surface type

<atom\_list> List of atoms for which the <distance> is to apply. The list can be the "all" keyword or a list of atom names.

<distance> Distance in Angstroem between an atom and the surface. If set to zero, the flag is not set.

Instead of the aom list, the values can be set for all atoms to:

"default" A default value of 2.55 Angstroem for all atoms.

"off" A value of zero Angstroem. No surface property flag will be set.

## show

### show

Show the current surface related settings.

## A.49 switch

**switch** <atom1>,<atom2>

**switch** <mole1>,<mole2>,"mol"



The atoms number <atom1> and <atom2> switch position. Internally, the chemistry of the two atoms is switched, while the positions remain as before. The corresponding properties are switched as well.

If 'mol' is given as a third parameter, molecules <mole1> and <mole2> are interchanged. The origin of the two molecules (position of atom 1) are unchanged.

## A.50 symmetry

### symmetry

Switches to the symmetry submenu of DISCUS. You can define a general symmetry operation in triclinic space such as a rotation around a general axis by a general angle. DISCUS performs a symmetry operation in general triclinic space according to the matrix notation:

$$\mathbf{r}' = \mathbf{S} \cdot \mathbf{r} + \mathbf{t}$$

Here,  $\mathbf{r}$  is the original position,  $\mathbf{r}'$  its image,  $\mathbf{S}$  the symmetry matrix and  $\mathbf{t}$  the translation vector. There are no restrictions on the direction of the symmetry axis, the rotation angle or the translational part. You can use this feature to create a twinned crystal, to rotate a group of atoms around a general axis etc.

Further help topics are:

### commands

Valid commands at this level are

```
@      ! Execute a macro file (see main help)
=      ! assigns the value to a variable (see main help)
angle  ! sets the angle of a rotation axis
asym   ! Shows asymmetric unit
calc   ! Calculates the symmetry transformation for a single point
chem   ! Shows the atoms present in the crystal
continue! Coninue a stopped macro (see main help level)
des     ! deselects atoms
echo   ! echo a string (see main help)
eval   ! Evaluates an expression for interactive check (see main help)
exit   ! terminates 'symm'
help   ! gives on line help for symmetry operations (see main help)
hkl    ! sets the symmetry axis in reciprocal space coordinates
incl   ! sets the range of atoms to be included in the symmetry operation
mdes   ! deselects molecules
mincl  ! sets the range of molecules to be included in the
        ! symmetry operation
mode   ! allows choice of creating a copy or replacing the atom
msel   ! selects molecules to be included in the symmetry operations
odes   ! deselects objects
oincl  ! sets the range of objects to be included in the symmetry operation
origin ! sets the origin for the symmetry operation
osel   ! selects objects to be included in the symmetry operations
power  ! sets the power of the symmetry operation
run    ! starts the calculation of the symmetry operations
sel    ! selects atoms to be included in the symmetry operations
show   ! shows the current parameters
stop   ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
```

```
trans    ! sets the translational part of the symmetry operation
type     ! Allows choice between proper and improper rotations
uvw      ! sets the symmetry axis in direct space coordinates
wait     ! Waits for user input (see main help)
```

## angle

**angle** <value>

Sets the angle of the symmetry operation. The angle must be given in degrees. No limit applies to the numerical value.

## asym

**asym**

Shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the variable `m[<index>]`. If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

## calc

**calc** *x,y,z* [*"d"* | *"r"* ]

Calculates the symmetry transformation for a single point *<x,y,z>*. The current settings for the symmetry operation are used to transform the vector *<x,y,z>*. The result is displayed and stored in the result array *"res"*. By default, or if the optional fourth parameter is set to *"d"*, DISCUS assumes that the vector is a real space vector. If the optional fourth parameter is set to *"r"* the vector is interpreted as a reciprocal space vector and the corresponding symmetry matrix in reciprocal space is used.

## chem

**chem**

Displays the type of atoms present in the crystal. For each type of atom, its scattering curve number, its name and its temperature factor are listed. Warning, even, if all atoms of a particular type have been deleted, its scattering type will remain in the list. This list could therefore include more types of atoms than are actually present in the crystal.

## des

**des** { *"all"* | *<name>* | *<number>* } [ , { *<name>* | *<number>* } ... ]

Deselects choices made by `==> 'sel'`. Possible values for the parameter are mutually exclusively:

```

"all"      all atoms of the crystal are deselected.
<name>    all the atoms called <name> of the crystal are deselected.
           This includes symmetrically not equivalent atoms.
<number>  all atoms of the crystal that are of scattering type <number>
           are deselected.

```

More than one atom may be deselected at once.

## hkl

**hkl** <h>,<k>,<l>

Specifies the direction of the symmetry axis in reciprocal space coordinates. The corresponding direct space coordinates are calculated by the program. Alternatively you may specify the direction in direct space by the ==> 'uvw' command.

## incl

**incl** {<start>,<end> | "all" | "envi" }

The symmetry operation includes all atoms numbered <start> to <end> inclusively. All other atoms are ignored. If, instead of explicit numbers, the parameter "all" is given, the symmetry operation will include all atoms of the crystal. This holds even, if at a later time you include further atoms in the crystal. Thus, you can define a setup for the symmetry operations early in a lengthy macro, then modify the crystal and just run the symmetry operation later on. If the parameter "envi" is specified, the symmetry operation will include all atoms in the current environment that was located by the last ==> 'find env' command. In addition you can define the atoms that are affected by the symmetry operation with the ==> 'sele' and 'dese' commands.

## mdes

**mdes** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'msel'. Possible values for the parameter are mutually exclusively:

```

"all"      all molecules of the crystal are deselected.
<number>  all molecules of the crystal that are of type <number>
           are deselected.

```

More than one molecule type may be deselected at once.

## mincl

**mincl** {<start>,<end> | "all" }

The symmetry operation includes all molecules numbered <start> to <end> inclusively. All other molecules are ignored. If, instead of explicit numbers, the parameter "all" is given, the symmetry operation will include all molecules of the crystal. This holds even, if at a later time you include further molecules in the crystal. Thus, you can define a setup for the symmetry operations early in a lengthy macro, then modify the crystal and just run the symmetry operation later on. In addition you can define the molecules that are affected by the symmetry operation with the ==> 'msele' and 'mdese' commands.

**mode**

**mode** { "copy" | "repl" } [, { "new" | "old" } ]

The symmetry operation can have two different effects on the current atom. In the "copy" mode, the symmetry operation creates a copy of the current atom. In the "repl" mode, the symmetry operation moves the current atom from its old position to the new position.

The optional second parameter has only an effect if molecules are used. It determines if the new molecule created by the symmetry operation has is of the same type as the original ("old") or is treated as a new molecule type ("new").

**msel**

**msel** { "all" | <number> } [, <number> ...]

This command executes serves to select those molecules that will be modified by the symmetry operation.

Defines which molecules are included in symmetry operation. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all molecules of the crystal are included.
           This includes the empty molecules in the structure, which are
           stored as scattering curve number zero.
<number>  all molecules of the crystal that are of type <number>
           are included.
```

More than one atom may be selected at once.

The selection of molecules by 'msel' and atoms by 'sel' are mutually exclusive. The symmetry operation uses either atoms or molecules.

**odes**

**odes** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'osele'. Possible values for the parameter are mutually exclusively:

```
"all"      all objects of the crystal are deselected.
<number>  all objects of the crystal that are of type <number>
           are deselected.
```

More than one object type may be deselected at once.

**oincl**

**oincl** {<start>,<end> | "all" }

The symmetry operation includes all objects numbered <start> to <end> inclusively. All other objects are ignored. If, instead of explicit numbers, the parameter "all" is given, the symmetry operation will include all objects of the crystal. This holds even, if at a later time you include further objects in the crystal. Thus, you can define a setup for the symmetry operations early in a lengthy macro, then modify the crystal and just run the symmetry operation later on. In addition you can define the objects that are affected by the symmetry operation with the ==>'osele' and 'odese' commands.

**origin**

**origin** <x>,<y>,<z> [, { "mol" | "cryst" } ]

Specifies the origin of the symmetry operation in direct space coordinates. The symmetry axis intersects this point and has the direction specified by ==> 'hkl' or 'uvw'. If the symmetry operation is an improper rotation, the center of symmetry is at <x>,<y>,<z>.

In case of molecules to be transformed, the optional last parameter specifies if the given origin is relative to the origin of the molecule ("mol") or the crystal ("cryst"). Note that the first atom within a molecule has to be on the molecules origin (see manual).

**osel**

**osel** { "all" | <number> } [, <number> ...]

This command serves to select those objects that will be modified by the symmetry operation. Defines which objects are included in symmetry operation. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all objects of the crystal are included.
           This includes the empty objects in the structure, which are
           stored as scattering curve number zero.
<number>  all molecules of the crystal that are of type <number>
           are included.
```

More than one atom may be selected at once.

The selection of objects by 'osel' and atoms by 'sel' are mutually exclusive. The symmetry operation uses either atoms or objects.

**power**

**power** <n> {, "multiple" | "single" }

Sets the power of the symmetry operation. The symmetry operation is applied <n> times.

If the ==> 'mode' of the symmetry operation is 'repl', the second parameter is meaningless.

The symmetry operation that applies in this case is :

$$r' = s^n * r + t^n$$

The atom is moved to the new position that results from an <n> fold application of the symmetry operation.

If the ==> 'mode' of the symmetry operation is "copy", <n> new atoms are created. The position of the first atom results from a single application of the symmetry operation, the position of the second from a twofold application etc.

**run****run**

Starts the symmetry operation.

**sel**

**sel** { "all" | <name> | <number> } [, { <name> | <number> } ...]

This command serves to select those atoms that will be modified by the symmetry operation. Defines which atoms are included in symmetry operation. Possible values for the first mandatory parameter are mutually exclusively:

"all"      all atoms of the crystal are included.

This includes the "voids" in the structure, which are stored as scattering curve number zero.

<name>      all the atoms called <name> of the crystal are included.  
This includes symmetrically not equivalent atoms.

<number>   all atoms of the crystal that are of scattering type <number>  
are included.

More than one atom may be selected at once.

The selection made stay valid until explicitly deselected!

**show****show**

Shows the current parameters of the symmetry operation.

**trans**

**trans** <t1>,<t2>,<t3>

Sets the translational part of the symmetry operation. The translation is applied after the rotation according to the matrix notation:

$$\mathbf{r}' = \mathbf{S} \cdot \mathbf{r} + \mathbf{t}$$

Here,  $\mathbf{r}$  is the original position,  $\mathbf{r}'$  its image,  $\mathbf{S}$  the symmetry matrix and  $\mathbf{t}$  the translation vector.

**type**

**type** {"proper" | "improper" }

A general rotation may be either a "proper" or an "improper" rotation. An improper rotation is a normal rotation followed by an inversion. To create, for example, a mirror operation specify a 180 degree angle and an improper rotation.

**uvw**

**uvw** <h>,<k>,<l>

Specifies the direction of the symmetry axis in direct space coordinates. The corresponding reciprocal space coordinates are calculated by the program. Alternatively you may specify the direction in reciprocal space by the ==> 'hkl' command.

terminates 'wave' returns to the normal discus prompt

## A.51 transform

### tran

Switches to the unit cell transformation submenu of DISCUS. At this sublevel you can define the relationship between an old and a new unit cell and perform the transformation of the atoms in the crystal. An interactive transformation allows to calculate the result for any single real and reciprocal space vector in both directions: old ==> new and new ==> old.

You can specify the relationship between the two unit cells in any of four possible options: define the new base vectors a,b,c in terms of the old base define the old base vectors a,b,c in terms of the new base define the new coordinates x,y,z in terms of the old coordinates define the old coordinates x,y,z in terms of the new coordinates define the new base vectors a\*,b\*,c\* in terms of the old reciprocal base define the old base vectors a\*,b\*,c\* in terms of the new reciprocal base

Independent of the choice above, you can define an optional shift of the origin by: defining the coordinates of the new origin in terms of the old base defining the coordinates of the old origin in terms of the new base

If all atoms in the crystal are transformed to the new base vectors, then the unit cell dimensions and the metric tensors are transformed as well. The space group is set to "P1" to prevent erroneous symmetry operations once the present crystal is saved to file and read again. All original symmetry operations are transformed and stored as additional generators.

### commands

Valid commands at this level are

```
@      ! Execute a macro file (see main help)
=      ! assigns the value to a variable (see main help)
anew   ! sets the new base vector "a" in terms of the old base
aold   ! sets the old base vector "a" in terms of the new base
asnew  ! sets the new reciprocal base vector "a" in terms of the old base
asold  ! sets the old reciprocal base vector "a" in terms of the new base
asym   ! Shows asymmetric unit
bnew   ! sets the new base vector "b" in terms of the old base
bold   ! sets the old base vector "b" in terms of the new base
bsnew  ! sets the new reciprocal base vector "b" in terms of the old base
bsold  ! sets the old reciprocal base vector "b" in terms of the new base
c2new  ! Calculates the transformation to "new" for a single vector
c2old  ! Calculates the transformation to "old" for a single vector
chem   ! Shows the atoms present in the crystal
cnew   ! sets the new base vector "c" in terms of the old base
cold   ! sets the old base vector "c" in terms of the new base
csnew  ! sets the new reciprocal base vector "c" in terms of the old base
csold  ! sets the old reciprocal base vector "c" in terms of the new base
continue! Coninue a stopped macro (see main help level)
des    ! deselects atoms
echo   ! echo a string (see main help)
eval   ! Evaluates an expression for interactive check (see main help)
exit   ! terminates 'tran'
help   ! gives on line help for unit cell transformations (see main help)
h2new  ! transforms a list of reflections to the new base system
h2old  ! transforms a list of reflections to the old base system
incl   ! sets the range of atoms to be included in the transformation
onew   ! sets the new origin in terms of the old base
old    ! sets the old origin in terms of the new base
```

```

run      ! starts the transformation for the selected atoms
sel      ! selects atoms to be included in the transformation
set      ! sets parameters
show     ! shows the current parameters
stop     ! Stops execution of a macro (see main help level)
system   ! Executes operating system command (see main help)
wait     ! Waits for user input (see main help)
xnew     ! sets the new coordinate "x" in terms of the old x,y,z
xold     ! sets the old coordinate "x" in terms of the new x,y,z
ynew     ! sets the new coordinate "y" in terms of the old x,y,z
yold     ! sets the old coordinate "y" in terms of the new x,y,z
znew     ! sets the new coordinate "z" in terms of the old x,y,z
zold     ! sets the old coordinate "z" in terms of the new x,y,z

```

**anew****anew a,b,c**

Defines the new base vector "a" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the old base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'anew', 'bnew' and 'cnew'.

**aold****aold a,b,c**

Defines the old base vector "a" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the new base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'aold', 'bold' and 'cold'.

**asnew****asnew a,b,c**

Defines the new reciprocal base vector "a" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the old reciprocal base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'asnew', 'bsnew' and 'csnew'.

**asold****asold a,b,c**

Defines the old reciprocal base vector "a" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the new reciprocal base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'asold', 'bsold' and 'csold'.

**asym****asym**

Shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the



variable  $m[\langle \text{index} \rangle]$ . If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

## **bnew**

### **bnew a,b,c**

Defines the new base vector "b" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the old base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'anew', 'bnew' and 'cnew'.

## **bold**

### **bold a,b,c**

Defines the old base vector "b" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the new base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'aold', 'bold' and 'cold'.

## **bsnew**

### **bsnew a,b,c**

Defines the new reciprocal base vector "b" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the old reciprocal base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'asnew', 'bsnew' and 'csnew'.

## **bsold**

### **bsold a,b,c**

Defines the old reciprocal base vector "b" in terms of multiples  $\langle a \rangle$ ,  $\langle b \rangle$ ,  $\langle c \rangle$  of the new reciprocal base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'asold', 'bsold' and 'csold'.

## **c2new**

### **c2new x,y,z [, ["d" | "r" ] ]**

Calculates the transformation from the old coordinate system to the new system for a single vector  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$ . Default is a direct space vector, the fourth optional parameter allows you to define  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$  as a reciprocal space vector. The result of the transformation is displayed on the screen and stored in the first three elements of the result array "res[i]".

## **c2old**

### **c2old x,y,z [, ["d" | "r" ] ]**

Calculates the transformation from the new coordinate system to the old system for a single vector  $\langle x \rangle, \langle y \rangle, \langle z \rangle$ . Default is a direct space vector, the fourth optional parameter allows you to define  $\langle x \rangle, \langle y \rangle, \langle z \rangle$  as a reciprocal space vector. The result of the transformation is displayed on the screen and stored in the first three elements of the result array "res[i]".

**csnew****csnew a,b,c**

Defines the new reciprocal base vector "c" in terms of multiples  $\langle a \rangle, \langle b \rangle, \langle c \rangle$  of the old reciprocal base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'asnew', 'bsnew' and 'csnew'.

**csold****csold a,b,c**

Defines the old reciprocal base vector "c" in terms of multiples  $\langle a \rangle, \langle b \rangle, \langle c \rangle$  of the new reciprocal base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'asold', 'bsold' and 'csold'.

**chem****chem**

Displays the type of atoms present in the crystal. For each type of atom, its scattering curve number, its name and its temperature factor are listed. Warning, even, if all atoms of a particular type have been deleted, its scattering type will remain in the list. This list could therefore include more types of atoms than are actually present in the crystal.

**cnew****cnew a,b,c**

Defines the new base vector "c" in terms of multiples  $\langle a \rangle, \langle b \rangle, \langle c \rangle$  of the old base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'anew', 'bnew' and 'cnew'.

**cold****cold a,b,c**

Defines the old base vector "c" in terms of multiples  $\langle a \rangle, \langle b \rangle, \langle c \rangle$  of the new base vectors. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'aold', 'bold' and 'cold'.

**des**

**des { "all" |  $\langle \text{name} \rangle$  |  $\langle \text{number} \rangle$  } [, {  $\langle \text{name} \rangle$  |  $\langle \text{number} \rangle$  } ...]**

Deselects choices made by ==> 'sel'. Possible values for the parameter are mutually exclusively:

```
"all"      all atoms of the crystal are deselected.
<name>     all the atoms called <name> of the crystal are deselected.
            This includes symmetrically not equivalent atoms.
<number>   all atoms of the crystal that are of scattering type <number>
            are deselected.
```

More than one atom may be deselected at once.

## h2new

### h2new <infile>

Transforms the reflections read from <infile> into reflections according to the new base system. The input format is the SHEXL format(3i4,2f8.2,i4). All new reflections that are integer are written to the outputfile named <infile>".trans" in SHELX format. You can set an accuracy for being integer by the ==> 'set deltahkl' command. All other reflections, that are more than +- <deltahkl> away from integer h,k,l are written to the file <infile>".rest" as hkl(new), intensity, sigma, batch, hkl(old). Here hkl(new) are real values.

## h2old

### h2old <infile>

Transforms the reflections read from <infile> into reflections according to the old base system. The input format is the SHEXL format(3i4,2f8.2,i4). All old reflections that are integer are written to the outputfile named <infile>".trans" in SHELX format. You can set an accuracy for being integer by the ==> 'set deltahkl' command. All other reflections, that are more than +- <deltahkl> away from integer h,k,l are written to the file <infile>".rest" as hkl(old), intensity, sigma, batch, hkl(new). Here hkl(old) are real values.

## incl

**incl { <start>,<end> | "all" }**

The unit cell transformation includes all atoms numbered <start> to <end> inclusively. All other atoms are ignored. If, instead of explicit numbers, the parameter "all" is given, the unit cell transformation will include all atoms of the crystal. This holds even, if at a later time you include further atoms in the crystal. Thus, you can define a setup for the unit cell transformations early in a lengthy macro, then modify the crystal and just run the unit cell transformation later on. In addition you can define the atoms that are affected by the unit cell transformation operation with the ==>'sele' and 'dese' commands.

## onew

### onew a,b,c

Defines the position of the new origin in terms of multiples <a>,<b>,<c> of the old base vectors. The default at program startup is 0.0, 0.0, 0.0

**oold****oold a,b,c**

Defines the position of the old origin in terms of multiples  $\langle a \rangle, \langle b \rangle, \langle c \rangle$  of the new base vectors. The default at program startup is 0.0, 0.0, 0.0

**run****run**

Starts the transformation operation.

**sel**

**sel { "all" |  $\langle \text{name} \rangle$  |  $\langle \text{number} \rangle$  } [, {  $\langle \text{name} \rangle$  |  $\langle \text{number} \rangle$  } ...]**

This command serves to select those atoms that will be modified by the unit cell transformation. Defines which atoms are included in unit cell transformations. Possible values for the first mandatory parameter are mutually exclusively:

**"all"**      all atoms of the crystal are included.

This includes the "voids" in the structure, which are stored as scattering curve number zero.

**$\langle \text{name} \rangle$**       all the atoms called  $\langle \text{name} \rangle$  of the crystal are included.  
This includes symmetrically not equivalent atoms.

**$\langle \text{number} \rangle$**       all atoms of the crystal that are of scattering type  $\langle \text{number} \rangle$   
are included.

More than one atom may be selected at once.

The selection made stay valid until explicitly deselected!

**set****set "deltahkl", $\langle \text{value} \rangle$** 

Sets the value of deltahkl to  $\langle \text{value} \rangle$ , as needed for  $\Rightarrow$  'h2new' and 'h2old'. When DISCUS transforms a list of reflections, it checks, how close the transformed values are to integer values h,k,l. The allowed deviation from integer numbers is set by this command.

**show****show**

Shows the current parameters of the transformation operation.

**xnew****xnew a,b,c**

Defines the transformation through the relationship between the new "x" coordinate of an atom in terms of multiples  $\langle x \rangle, \langle y \rangle, \langle z \rangle$  of the old coordinates of the atom. If you choose this definition of the unit cell transformation, you MUST define all three  $\Rightarrow$  'xnew', 'ynew' and 'znew'.

**xold****xold a,b,c**

Defines the transformation through the relationship between the old "x" coordinate of an atom in terms of multiples  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$  of the new coordinates of the atom. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'xold','yold' and 'zold'.

**ynew****ynew a,b,c**

Defines the transformation through the relationship between the new "y" coordinate of an atom in terms of multiples  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$  of the old coordinates of the atom. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'xnew','ynew' and 'znew'.

**yold****yold a,b,c**

Defines the transformation through the relationship between the old "y" coordinate of an atom in terms of multiples  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$  of the new coordinates of the atom. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'xold','yold' and 'zold'.

**znew****znew a,b,c**

Defines the transformation through the relationship between the new "z" coordinate of an atom in terms of multiples  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$  of the old coordinates of the atom. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'xnew','ynew' and 'znew'.

**zold****zold a,b,c**

Defines the transformation through the relationship between the old "z" coordinate of an atom in terms of multiples  $\langle x \rangle$ ,  $\langle y \rangle$ ,  $\langle z \rangle$  of the new coordinates of the atom. If you choose this definition of the unit cell transformation, you MUST define all three ==> 'xold','yold' and 'zold'.

**A.52 ther**

```
ther [{"all" | "2d"}]
ther "mol", [{"all" | "2d"}]
```

Randomly displaces all atoms according to their respective isotropic thermal coefficient. The directions of the displacements are distributed in uniform random distribution, the amplitude of the displacements are Gaussian distributed with FWHM calculated from the isotropic B. As default the displacements are performed in all 3 dimensions. For 2d simulations however, the displacement in all directions corresponding to a size of one unit cell can be switched off by the parameter "2d". The value "all" is equal to the default.

If the second parameter is set to "mol", the complete rigid molecule is displaced according to the isotropic temperature factor defines via a "mole\_biso[i]=" statement or specified on a "molecule biso,<value>" statement in the ==> Unit cell file.

The temperature factor of the atom at the origin of the molecule is NO longer used!

## A.53 variables

The program recognizes free variables:

```
"i[ ]"  integer variables
"r[ ]"  real variables
```

The current implementation allows 501 free variables [0] through [501] each.

```
"res[ ]"  Result of several calculations that return more than one
          value.
          res[0] contains the number of values returned, res[1] to
          res[res[0]] contain the actual values.
          The values of "res" change every time, a command is applied
          that returns more than one value. The old values are lost !
```

Several other variables are linked to structural values:

```
"x[<i>]" ,
"y[<i>]" and
"z[<i>]" are the fractional coordinates of the atom number <i>.
"at_name[<i>]" is a character variable that hold the name of
              of atom number <i>
"m[<i>]" is the number of the scattering curve for atom <i>.
"b[<t>]" is the isotropic B value for atom type <t>.
"n[1]"   is the total number of atoms in the crystal, this variable
          is read only.
"n[2]"   is the total number of different scattering curves in the
          crystal, this variable is read only.
"n[3]"   is the total number of atoms in the original unit cell,
          this variable is read only.
"n[4]"   is the total number of molecules in the crystal, this variable
          is read only.
"n[5]"   is the total number of different molecule types in the
          crystal, this variable is read only.
"n[6]"   is the total number of molecules in the original unit cell,
          this variable is read only.
"n[7]"   is the total number of non-void atoms in the original unit cell,
          this variable is read only.
"cdim[1,1]" lowest x coordinate of any atom in the crystal.
"cdim[1,2]" highest x coordinate of any atom in the crystal.
"cdim[2,1]" lowest y coordinate of any atom in the crystal.
"cdim[2,2]" highest y coordinate of any atom in the crystal.
"cdim[3,1]" lowest z coordinate of any atom in the crystal.
"cdim[3,2]" highest z coordinate of any atom in the crystal.
```

The variable "cdim" is read only.

```
"lat[<i>]"  Lattice constants a,b,c,alpha,beta,gamma
"vol[1]"    Unit cell volume in A**3
"rvol[1]"   Reciprocal unit cell volume in A**(-3)
"env[<i>]"   Index of neighbouring atoms as found by
              ==> "find"
              "env[0]" is the number of neighbours.
"menv[<i>]"   Index of neighbouring molecules as found by
              ==> "find"
              "menv[0]" is the number of neighbours.
"mol_cont[i1,i2]" Content of molecule number i1.
                  mol_cont[i1,0] gives the number of
                  atoms in the molecule. For <i2> > 0, the
                  information about the <i2>'nd atom in the
                  molecule is given. The number of the
                  atom in the crystal is returned.
                  read only
"mol_len[i1]"  Number of atoms in molecule i1.
               read only
"mole_type[i1]" Type of molecule no. i1
               read only
"mole_biso[i1]" Isotropic B-value for Molecule i1
               read/write
"in_mole[i1]"  Molecule number in which atom i1 is located
               read only
"pdf_dens[0]"  Number density for PDF calculations
"pdf_scal[0]"  Scale factor for PDF calculations
```

The variables can be assigned constant numbers:

```
i[0] = 1           ! Integer variable no 0 is set to "1"
r[4] = 1.5         ! Real variable no 4 is set to "1.5"
x[5] = -2.7888     ! x-coordinate of atom 5 is set to "-2.7888"
```

Arithmetic expressions may be used on the right hand side:

```
i[i[4]+2] = 2.0*(i[2]+i[0])/(i[2]**2)
m[i[0]+50] = m[i[0]] ! The atom 50 atoms down the line from
                     atom no <i[0]> is assigned the same
                     scattering curve, thus essentially switching
                     its chemistry around !
```

As shown in the example above, arithmetic operations can be nested and grouped by brackets "(" and ")".

## A.54 vprod

**vprod <u1>,<v1>,<w1>,<u2>,<v2>,<w2> [,<flag>]**

This command calculates the vector product  $\langle uvw\ 1 \rangle \times \langle uvw\ 2 \rangle$ . The <flag> is a string of three characters that indicates whether the input vectors or the output vector are given in direct or reciprocal space coordinates. The three characters correspond to:

```
    first input vector,
    second input vector
    resulting vector
"drd" means: vector one is given in direct space coordinates, vector 2
              is in reciprocal space coordinates (hkl) and the
              resulting vector product is to be given in direct space
              coordinates
```

Any combination of "d" and "r" is allowed.

The output vector is stored in the result variable: "res[1]","res[2]","res[3]" vector product of vector 1 and vector 2

## A.55 waves

### wave

displaces the atoms along a wave. It generates sublevel with prompt "discus/wave >". Further help topics are:

### commands

Valid commands at this level are

```
@      ! Execute a macro file (see main help)
=      ! Assigns the value to a variable (see main help)
acco   ! Sets the wave type to acoustic
amp    ! Sets the amplitude in Angstrom
asym   ! Shows asymmetric unit
chem   ! Shows the atoms present in the crystal
continue! Coninue a stopped macro (see main help level)
dens   ! Sets the wave type to density wave
des    ! Deselects atoms
echo   ! Echo a string (see main help)
eval   ! Evaluates an expression for interactive check (see main help)
exit   ! Terminates 'wave'
func   ! Sets the wave function
help   ! Gives on line help for waves (see main help)
len    ! Sets the wave length in Angstrom
long   ! Sets the wave type to longitudinal
mdese  ! Deselects molecule types
mrepl  ! Selects molecules to be replaced by density wave
msel   ! Selects molecule types
opti   ! Sets the wave type to optical
osci   ! Oscillation vector for transverse waves
phase  ! Sets the phase of the wave at 0,0,0 in direct space
phigh  ! Sets the high probability for density waves
plow   ! Sets the low probability for density waves
repl   ! Selects atoms to be replaced by a density wave
rot    ! Sets type to rotational wave (molecules only)
run    ! Starts the calculation of the wave
sel    ! Selects atoms to be included in the wave
shift  ! Allows a constant shift to be added to the displacement
show   ! Show current settings
stop   ! Stops execution of a macro (see main help level)
system ! Executes operating system command (see main help)
tran   ! Sets the wave type to transverse
vect   ! Sets wave vector
wait   ! Waits for user input (see main help)
```

### acco

### acco

Sets the wave type to acoustic, atoms of opposite charge are displaced in the same direction.



**amp****amp** <value>

sets the amplitude in Angstrom

**asym****asym**

Shows the content of the asymmetric unit. The names of those atoms, a number that is used as index for its scattering type, their position and temperature coefficient are listed. The number that is listed, is the number that refers to the scattering curve of that atom. It is contained in the variable `m[<index>]`. If a cell was read, all atoms are considered to be different, even if they are chemically identical and have the same temperature coefficient. If a whole structure was read, all atoms that are in the unit cell  $0 \leq xyz < 1$ , are chemically unique and have a different temperature coefficient are included in the asymmetric unit.

**chem****chem**

Displays the type of atoms present in the crystal. For each type of atom, its scattering curve number, its name and its temperature factor are listed. Warning, even, if all atoms of a particular type have been deleted, its scattering type will remain in the list. This list could therefore include more types of atoms than are actually present in the crystal.

**dens****dens**

Sets the type of the wave to be a density wave. All atoms selected by 'sel' or 'repl' will be modulated by the wave. The probability with which an atom is present oscillates between 'plow' and 'phigh'.

**des****des** { "all" | <name> | <number> } [, { <name> | <number> } ...]

Deselects choices made by ==> 'sel' or ==> 'repl'. Possible values for the parameter are mutually exclusively:

```
"all"      all atoms of the crystal are deselected.
<name>     all the atoms called <name> of the crystal are deselected.
            This includes symmetrically not equivalent atoms.
<number>   all atoms of the crystal that are of scattering type <number>
            are deselected.
```

More than one atom may be deselected at once.

**func****func "box" [,<delta>]****func "sinus"****func "triang"**

Sets the type of wave function used. The parameter can be any of the strings listed.

A) displacement waves:

```
"box"      : A box shaped wave function is used. The function is
              symmetrical around the origin. This, however, can be
              changed by changing the phase at the origin, see command
              ==> 'phase'. The default is a symmetrical box function,
              i.e. a value of <delta> of 0.5. The value of <delta>
              can be given as additional parameter to us an asymmetric
              box function, i.e. maxima and minima have different widths.
"sinus"    : A sinusoidal wave function is used. The displacement is given
              by amp*SIN(arg).
"triang"    : A triangular displacement is applied. The displacement linearly
              increases from zero to amp and drops back to zero at the end
              of the wave. The point at 0/4 = 4/4 wave length has displacement
              zero.
```

The constant shift is added to these displacements.

B) replacement waves:

```
"box"      : A box shaped wave function is used. The occupation probability
              of the original atom, see ==> 'repl', is equal to <plow> along
              half the wave length and <phigh> else. Other details were
              given above.
"sinus"    : A sinusoidal wave function is used. The occupation probability
              is given by
              amp * SIN(arg) + amp0, where
              amp = 0.5*( phigh-plow)
              amp0= 0.5*( phigh+plow)
              This will give a sinusoidal wave with minimum at <plow> and
              maximum value at <phigh>.
"triang"    : A triangular occupation probability is applied. The occupation
              probability linearly increases from <plow> to <phigh> and drops
              back to zero at the end of the wave.
              The point at 0/4 = 4/4 wave length has the occupation
              probability of <plow>.
```

**len****len <value>**

Sets the wave length in Angstrom

**long****long**

Sets the wave type to longitudinal

**mdes**

**mdes** { "all" | <number> } [, <number> ...]

This command deselects all or individual molecule types used for the plotting output.

**msel**

**msel** { "all" | <number> } [, <number> ...]

This command allows to select the molecule types to be used for the plotting output. By using 'msel' the atom selection becomes invalid. The parameter "all" will select ALL molecule types. Alternatively individual molecule types can be selected using the corresponding <number>. In order to use atoms again, use the 'sele' command.

**mrep**

**mrep** <typ1>,<typ2> [,"viceversa"]

This command selects the molecules that will be replaced by a density function. The parameter <typ1> specifies the molecule type to eventually be replaced by type <typ2>. Repeat the command to select more than one molecule type. To deselect use 'mdes'. To remove the molecules rather than replacing them, set <typ2> to zero.

If the third optional parameter "viceversa" is given, the molecules of type <typ2> are replaced by molecules of type <typ1> with the opposite probability. This is useful, if you have a random arrangement of molecule types <typ1> and <typ2> and would like to obtain a density wave that fluctuates between these two types. If the optional parameter is omitted, only molecules of type <typ1> are replaced and all molecules of type <typ2> are not changed. Keep in mind that you should select only those molecules that you want to exchange, and do not use the ==> msel all construction.

**rot**

**rot** <u>,<v>,<w> [,<x>,<y>,<z>]

This command selects the rotational wave mode which is only available when using molecules. The wave amplitude is here used as rotation angle rather than as displacement or probability as for the other modes. The parameters <u>,<v>,<w> define the rotation axis in real space. The origin of the rotation axis is the origin of the molecule. However, an offset <x>,<y>,<z> for this origin can be specified as optional parameters.

**opti**

**opti**

Sets the wave type to optical, atoms of opposite charge are displaced in opposite directions.

**osci**

**osci** <x,y,z>

Sets the direction of the oscillation vector for transverse waves.

### phase

**phase** { <value> | "random" }

Sets the phase of the wave at 0,0,0 in direct space in degrees. The initial phase is zero. With phase = 0, a sine wave results, with phase = 90 a cosine wave.

### phigh

**phigh** <value>

Sets the upper probability limit, with which an atom is retained by a density wave. The maximum occupancy of the atom reaches <value>.

### plow

**plow** <value>

Sets the lower probability limit, with which an atom is retained by a density wave. The minimum occupancy of the atom reaches <value>.

### repl

**repl** { "all" | <name> | <number> }, { <name> | <number> } [,"viceversa"]

This command selects those atoms that will be replaced by a density function. Possible values for the first mandatory parameter are mutually exclusively:

```
"all"      all atoms of the crystal are replaced by the atom defined
            by the second parameter.
<name>     all the atoms called <name> of the crystal are replaced by
            the atom defined by the second parameter. This includes
            symmetrically not equivalent atoms.
<number>   all atoms of the crystal that are of scattering type <name>
            are replaced by the atom defined by the second parameter.
```

The second parameter defines the atom, the original is replaced by. If a name is given, it need not to be present in the original crystal. A new scattering type is automatically added to the table. If a number is given, to which no corresponding atom exists, an error message is displayed.

If the third optional parameter "viceversa" is given, the atoms of the first type are replaced by atoms of the second type with the opposite probability. This is useful, if you have a random arrangement of atoms types and would like to obtain a density wave that fluctuates between these two types. If the optional parameter is omitted, only atoms of the first type are replaced and all atoms of the second type are not changed. Keep in mind that you should select only those atoms that you want to exchange, and do not use the ==> sel all construction.

**run****run**

Starts the calculation of the wave

**sel**

**sel** { "all" | <name> | <number> } [, { <name> | <number> } ...]

This command serves to select those atoms that will be modified by a wave function. Defines which atoms are included in a wave. Possible values for the first mandatory parameter are mutually exclusively:

**"all"**      all atoms of the crystal are included.

This includes the "voids" in the structure, which are stored as scattering curve number zero.

**<name>**      all the atoms called <name> of the crystal are included.

              This includes symmetrically not equivalent atoms.

**<number>**   all atoms of the crystal that are of scattering type <number> are included.

More than one atom may be selected at once.

The selection made stay valid until explicitly deselected!

**shift**

**shift** <amount>

A constant shift <amount> is added to the displacement of the atoms

**show****show**

shows the current settings for the wave.

**tran****tran**

sets the wave type to transverse

**vect**

**vect** <x,y,z>

sets the direction of the wave vector in units of the lattice constants. The wave length has to be set with ==> "len".

## A.56 wyckoff

**wyckoff** <x>,<y>,<z> [,{"full" | "symbol" | "xyz" | "matrix" }]

Shows all symmetry operations that copy an atom at position <x>,<y>,<z> onto itself. These are the symmetry operations of the current space group, not the simple point group symmetry.

"full"

The default setting, produces a full listing. This includes: the symmetry matrix as (3x4) matrix the symbol as printed in the International Tables the resulting atom positions xyz, as printed in the International Tables

For each symmetry operation DISCUS lists the number in square brackets, and for centered space groups the number within each block that corresponds to one centering vector in round brackets. With the setting ==> set generator,symmetry the sequence corresponds to that listed in the Tables.

"symbol"

This produces a short list that consists of the symbol as used in the International Tables.

"xyz"

This produces a short list that consists of the symbolic atom positions xyz, as used in the International Tables.

"matrix"

This produces a list that consists of the (3x4) symmetry matrices.

## A.57 bugs

### Known bugs

Here is a list of known bugs of DISCUS which are on our 'repair list':

- DISCUS hangs if the input is redirected from a file ('discus < file') and the prompt status is ON if command line editing is used. Use 'set prompt,off' or 'set prompt,redirect' in the first line of the file to be redirected to avoid the problem.

## A.58 domain

This command branches to the domain menu level.

Domains are understood as any group of atoms which is to replace part of the original structure. A domain may be encoded within the file in two different ways: A) A list of pseudoatoms B) A list of explicit domains

The pseudoatom file structure is identical to any DISCUS structure file. The atoms are interpreted as positions of the domains within the host structure. Size, character, and orientation of the domain can be set by the user via the 'assign' commands at the domain level.

The domain lists encode the size, character, and orientation within the file itself.

Valid commands at this sublevel are:

## commands

```

assign      ! Assigns size, character etc to the pseudoatoms
inputfile   ! Defines the input file name
mode        ! Defines the interpretation of the input file
run         ! Perform the actual replacement
set         ! Sets various definitions
show        ! Show current settings

```

## Instructions on usage

These are some quick hints on how to use the domain level.

An explicit domain is encoded similar to a molecule:

```

domain domain character, domain_fuzzy domain file, domain_a.stru domain fuzzy, 0.8 posi 0.0000
0.0000 0.0000 0.000 xaxi 1.0000 0.0000 0.0000 1.000 yaxi 0.0000 1.0000 0.0000 1.000 zaxi 0.0000
0.0000 1.0000 1.000 cent 0.0000 0.0000 0.0000 0.000 xdim 1.0000 0.0000 0.0000 1.000 ydim 0.0000
1.0000 0.0000 1.000 zdim 0.0000 0.0000 1.0000 1.000 domain end

```

This encoding allows you to use all of the DISCUS tools to create a distribution of domains within the host crystal. An alternative encoding is the use of "pseudoatoms". These are regular atoms, which are interpreted by the domain level as positions of a domain.

See the ==> 'insert' command for instructions to insert a domain into a dummy structure.

The domain character may be any of "domain\_cube", "domain\_cylinder", "domain\_fuzzy", or "domain\_sphere". A "domain\_cube" is a cube with dimensions given by the xdim, ydim, and zdim dimensions. These values are elements of a distortion matrix which allows you to rotate and distort the cube. A "domain\_cylinder" is a cylindrical domain. The initial rotation axis is parallel to the z-axis. A "domain\_fuzzy" domain has no predefined boundary. DISCUS reads all corresponding atoms from the file and deletes all host atoms, that are less than the fuzzy separation away from any of the domain atoms. A "domain\_sphere" is an initially spherical domain.

A domain replaces all atoms that exist prior within the crystal. The exception are those atoms with special names that define a domain or an object. These atoms are not replaced by a new domain. This allows you to replace a structure in part with other domains, where these domains remain intact, even if they overlap.

See the manual for full details.

The file contains the actual atoms that will be inserted into the host structure. You must make sure that the atoms from this file fill out the space freed by the domain.

The fuzzy value gives the minimum distance between domain atoms and host atoms. Any atom closer than this separation is deleted for the host structure. If the fuzzy distance is set to zero or a negative value, the check is not performed and all old host atoms remain in the structure.

The "posi" line is understood as the center of the domain atoms, the following three lines as the transformation matrix. The actual position onto which the domain atom is placed is calculated as:  $(t11 \ t12 \ t13) \cdot (xcoordinate) + (posi(x))$ ,  $(t21 \ t22 \ t23) \cdot (ycoordinate) + (posi(y))$ ,  $(t31 \ t32 \ t33) \cdot (zcoordinate) + (posi(z))$

Each of the matrix elements is calculated as follows:  $t11 = xaxi(x) - posi(x)$ ,  $t12 = xaxi(y) - posi(y)$ , ..  $t21 = yaxi(x) - posi(x)$  .. Thus the transformation matrix is actually encoded by adding the position to all matrix elements. This seemingly complication actually allows you to modify

the domain by all DISCUS elements like the generalized symmetry operation ==> 'symm'. A rotation for example will thus actually leave the matrix nicely intact.

Likewise the next four lines are used to encode the transformation matrix used to define the domain shape. Again, the actual transformation matrix is the difference between the lines "\*dim" and the "cent" line. By increasing the dimension, you increase the domain size. You can similaryl distort the domain shape.

The generalized symmetry level ==> "symm" allows you to change the position, orientation and shape of the domain and/or the atom transformation matrix.

### assign

```
assign "character",<pseudoatom name>,{"cube"|"cylinder"|"fuzzy"|"sphere"}
assign "content",<pseudoatom name>,<contentfile>
assign "fuzzy",<pseudoatom name>,<radius>
assign "orient",<pseudoatom name>,<row>,<ti1>,<ti2>,<ti3>,<ti4>
assign "shape",<pseudoatom name>,<row>,<ti1>,<ti2>,<ti3>,<ti4>
```

These commands allow you to assign extended properties to the pseudo atoms. Since the pseudo atoms are identical to regular atoms, they can only encode the position and type of a domain. The character, size, and orientation of the atoms with respect to the host structure may be assigned via these commands.

```
assign "character",<pseudoatom name>,{"cube"|"cylinder"|"fuzzy"|"sphere"}
```

A domain may be any of four different principal shapes. The "cube" domain is a domain consisting of the six {100} faces. The default distance of each face from the center is one respective unit cell. A "cylinder" domain has cylindrical geometry with the default cylinder axis parallel to the c-axis. The radius is one b-axis length. Top and bottom are terminated by the two {001} faces. A "sphere" domain initially is spherical with radius one b-axis.

All host atoms that are within the domain are removed, the void is filled by the atoms from the corresponding <contentfile>. The size and shape of these domains may be changed by adjusting the ==> 'assign shape' of the domain.

The "fuzzy" domain follows a very different concept. It does not have a predefined size or shape. Instead all atoms that are read from the <contentfile> replace host atoms that are close to their position.

```
assign "content",<pseudoatom name>,<contentfile>
```

The <contentfile> contains a DISCUS type structure file with all the atoms that are to be placed inside the host crystal. You need to make sure that the structure listed in the <contetnfile> is large enough to fill the empty void created by the respective domain. The space group and unit cell dimensions of the domain structure and the host crystal must be identical.

```
assign "fuzzy",<pseudoatom name>,<radius>
```

All host atoms that are closer than this <radius> to any of the domain atoms are removed from the original host structure. This holds for all domain character types. If the <radius> is set to zero or any negative number, the check is not performed for this domain type. Atoms of this domain type could thus be right on top of an atom of the host structure!

```
assign "orient",<pseudoatom name>,<row_i>,<ti1>,<ti2>,<ti3>,<ti4>
```



The orientation provides you a transformation matrix that is applied to each domain atom prior to insertion into the host. To keep the input lines short, each of the three rows <row\_i> must be given separately. Each atom is transformed by: (t11 t12 t13 ) (xcoordinate) (t14) (posi(x)) (t21 t22 t23 ) (ycoordinate) + (t24) + (posi(y)) (t31 t32 t33 ) (zcoordinate) (t34) (posi(z)) The translational part <ti4> is added to the position of the pseudo atom.

**assign "shape",<pseudoatom name>,<row>,<ti1>,<ti2>,<ti3>,<ti4>**

The shape transformation matrix is used to determine the size and shape of the domain. Default size at program start are +- one unit cell along each axis. Enlarge the domain by increasing the elements <tii> of the shape matrix, distort by changing the off-axis elements. The shape is distorted by: (t11 t12 t13 ) (xaxis) (t14) (posi(x)) (t21 t22 t23 ) (yaxis) + (t24) + (posi(y)) (t31 t32 t33 ) (zaxis) (t34) (posi(z)) The translational part <ti4> is added to the position of the pseudo atom. The shape transformation works independent from the atom transformation matrix. If the character of the domain is "cube", "cylinder", or "sphere", DISCUS checks whether the position of the domain atom after its transformation fits into the domain shape. If true, it is inserted into the guest structure.

## inputfile

**inputfile <inputfile>**

DISCUS read the domain positions from the file <inputfile>. See the ==> 'mode' command on how DISCUS interprets this file.

## mode

**mode {"domain" | "pseudoatoms"}**

This command tells DISCUS how to interpret the input file. With the "pseudoatoms" option, DISCUS expects to find a standard DISCUS structure file. Each "atom" in the file is interpreted as a domain position. See the ==> 'assign' commands for further details.

With the "domain" option DISCUS expects to read a file structured like: A general header like all DISCUS structure files followed by as many domain descriptors as needed. Each domain descriptor is designed similarly to the molecule and object descriptors:

## run

### run

This starts the actual incorporation of the domains into the host crystal. The positions, characters etc. for each domain are read from file <inputfile>. The corresponding domain structure is used to replace part of the host structure.

## set

### set

Set various definitions

**distance**

**set distance**, {"external" | "internal" | "all" }, <at\_list>, <distance>

Set the distances at which an atom from the atom list shall be considered to be close to a surface.

**remove**

**set remove** {"strict" | "initial" | "trust"}

When DISCUS replaces (parts of) an old structure with the content of domains, it needs to remove the old atoms. You have three choices how this shall be done:

**strict** Old atoms are removed EVERYTIME an individual domain is inserted into the crystal. Old atoms are considered to be all atoms that were in the crystal just before this domain was inserted. This includes domains that were inserted by the very same 'run' command. You need to do this, if you suspect or know that domains do overlap. You must use this for example to simulate overgrowth of domains on top of each other.

**initial** This time only those atoms are removed that were in the crystal prior to the current 'run' command. The atoms that are inserted during the current 'run' command are not tested. They may overlap if you are not careful.

**trust** OK, you trust that no domains overlap with each other and that you know exactly which host atoms have to be removed. The removal is performed after all domains have been inserted. DISCUS collects all fuzzy distances from the ==> assign commands and uses the maximum value of any assign command. This is necessary, since at this point the information from which domain type a new atom was taken has been lost. This removal is MUCH faster than the previous two.

**none** The ultimate trust! No host atoms are removed at all. You can do this, if you read a crystal that consisted only of dummy atoms and each of these represents a domain. You should remove these dummy atoms after you leave the 'domain' menu via a ==> replace dummy, void, all, 1.0 command. This combination is the fastest but be advised that you really need to know where the dummy atoms and the inserted atoms are. Otherwise you might end up with lots of strange overlapping atoms.

**show****show**

Shows the current settings.

## A.59 errors

The program has been written such that it should handle almost any typing error when giving commands and hopefully all errors that result from calculation with erroneous data. When an error is found an error message is displayed that should get you back on track. See the manual for a complete list of error messages.

Error messages concerning the command language are explained in the -> command language section of the online help. Application specific commands are described here and are grouped as follows:

```
APPL  Errors specific to DISCUS usage
CHEM  Errors at the chemistry sublevel
FOUR  Errors related to Fourier Transform
MMC   Errors at the Monte Carlo level
PDF   Errors at the PDF level
RMC   Errors at the Reverse Monte Carlo level
```

Each error message is displayed together with the corresponding category <cccc> and the error number <numb> in the form:

```
****CCCC****message **** numb ****
```

Type help error <cccc> <numb> to get an explanation for the error message and hints for possible steps to correct the situation.

In the default mode DISCUS returns the standard prompt and you can continue the execution from this point. You can set the error status to "exit" by the ==>'set' command. In this case DISCUS terminates if an error is detected. This option is useful to terminate a faulty sequence of commands when running DISCUS in the batch mode of your operating system. If you are a bit more daring, you can use the "live" error setting. With this setting, DISCUS will continue to run a macro after an error has occurred. This is helpful if you attempt to read a file of unknown length with ==> 'fget' and need to catch the EOF signal. Oter than that be careful!

## appl

This category lists error messages that are specific to DISCUS usage. You will find error messages regarding improper usage of atoms, space group violations, etc.

### Error -1: Maximum number of Orient. matrices read

The maximum number of orientation matrices that can be used for microdomains is limited by the parameter "MAX\_MD\_ORI" in the file "microdis.inc". If necessary adjust this parameter and recompile the program.

### Error -2: Improper limits for atom number

Either of the upper or lower limits used on the 'append' command is outside the range of atoms present in the crystal. Check whether the limits are both positive, the upper limit is higher or equal to the lower limit and whether both limits are less or equal to the number of atoms present in the crystal. The number of atoms in the crystal can be checked with the command: "eval n[1]".

### Error -3: No atoms selected yet

The plot and waves can only be run for selected atoms. Use the 'select' command to select individual atom types or to select all atoms present in the crystal.

### Error -4: Extend of plotspace is zero

The dimensions of the space from which atoms are to be plotted is zero. Check the values displayed by the plot 'show' command and use the 'exte' command to give appropriate values.

**Error -5: Too many microdomains within crystal**

DISCUS stores all microdomain positions in an array. The size of this array is defined by parameter MC\_MAXMICRO in file "microdis.inc". If necessary readjust the value of this parameter and recompile the program.

**Error -6: Unknown microdomain type**

The microdomain type number used with the 'place' command does not correspond to an existing microdomain type in the list. Check with 'show types' for a list of microdomain types that have been inserted into the list of available microdomains.

**Error -7: Unknown space group symbol**

The crystal file contains an unknown space group. Check the spelling of the space group symbol. The allowed space groups are all 230 space groups in the Int. Tables for Crystallography plus the space groups given for alternative settings and cell choices of the monoclinic space groups. Refer to appendix A of the manual for a complete listing of space group symbols.

**Error -8: Too many points in reciprocal layer**

The number of points in reciprocal space that can be calculated by the Fourier transform is limited by the parameter "MAXQXY" in the file "qspace.inc". Check that the product of <na> and <no> is less or equal to "MAXQXY". If necessary adjust the parameter "MAXQXY" and recompile the program.

**Error -9: Unknown Output Format**

An unknown output format was defined for the output level. Check that either of: "stan", "gnu", "pgm" or "post" was used as output format with the 'format' command.

**Error -10: Too many Atoms in crystal**

The maximum number of atoms that can be stored in the structure is determined by the parameter MAXATOMS in the file "crystal.inc". By inserting new atoms or by defining too large a crystal on the 'read' command, this number was exceeded. If necessary, change the value of the parameter MAXATOMS and recompile the program.

**Error -11: Unknown threshold type**

The parameter given on the 'thresh' command does not correspond to a valid entry. Check the spelling of the parameter.

**Error -12: Number of points must be > zero**

The value of the parameter given on the 'abs' or 'ord' command is less than one. This value represents the number of data points calculated along the respective direction. The value must be at least one or higher.

**Error -13: Correlation matrix index outside limits**

The index of the correlation matrix element(s) is either less than one or higher than the number of defined microdomain types. Check the 'ccol', 'crow' and 'cele' commands for proper number of parameters. Use the 'show type' command at 'microdomain' sub level to see the number of microdomains currently defined.

**Error -14: Invalid space group & lattice constants**

While reading a new cell the program checks the space group and the lattice constants for consistency. Either a space group was given that is not included in the program, or the lattice constants do not fulfill the constraints imposed by the space group. Check the space group symbol and the lattice constants given in the input file.

**Error -15: No microdomain input file name defined**

An attempt was made to insert a microdomain type without specifying the input file name. With the exception of content type 'free' all microdomains require an input file from which the new atom positions are read. Check the current defaults by 'show def' and the parameters given on then 'cont' command.

**Error -16: Status for log must be : "on" or "off"**

The second parameter for the 'log' command must be either "on" or "off".

**Error -17: Index of matrix outside limits**

The first parameter on the 'mrow' command is the index of the row that is to be inserted. Allowed values are one, two or three.

**Error -18: No orientation with this number exists**

The orientation matrix requested as parameter on the 'orie' command does not exist. Use the 'show orie' command to view the list of defined orientation matrices. Check whether three 'mrow' commands were used for each orientation matrix that you intended to set up.

**Error -19: Atom number outside limits**

The number of the atom is either less than one, larger than the current number of atoms in the crystal or even larger than the maximum number of atoms allowed in your implementation. Check the value of the parameter(s) on the 'remove' and 'switch' commands or check the index

of the variables "m", "x", "y" or "z". Check the number of atoms present in the crystal by the command: 'eval n[1]'.

**Error -20: Unknown element, no Fourier calculated**

An element was detected in the list of atoms for which there is no scattering curve available. The Fourier transform is not calculated at all. Check the name of all atoms present in the crystal using the 'asym' and 'chem' commands. If a charged ion was given, this valence might not be present in the list of scattering curves. Refer to Appendix b of the manual for a list of internally stored names. If the 'scat' and/or 'delf' commands were used, any name may be used. Check whether the commands were used properly.

**Error -21: No element present, no Fourier calculated**

There are no elements present at all in the crystal. The Fourier transform is not calculated at all. Most likely, the Fourier was called before a structure or unit cell was read, or an error occurred during the reading of the structure or unit cell.

**Error -22: Incompatible standard file sizes**

While attempting to add two Standard files, DISCUS encountered different length of rows or columns in the two input files. The adding is interrupted at this stage. The indices along the abscissa and ordinate at which the error occurred, is displayed as well, check the input file for erroneous content. Also check, whether the parameters nx,ny in the two input files are identical as well as the xmin,xmax,ymin and ymax parameters in the second line of both input files.

**Error -23: Incompatible GNUPLOT file sizes**

While attempting to add two GNUPLOT files, DISCUS encountered different length of rows or columns in the two input files. The adding is interrupted at this stage. The line number in which the error occurred, is displayed as well, check the input file for erroneous content.

**Error -24: Different coordinates in GNUPLOT files**

While attempting to add two GNUPLOT files, DISCUS encountered different coordinates in the two input files. The adding is interrupted at this stage. The line number in which the error occurred, is displayed as well, check the input file for erroneous content.

**Error -25: Different coordinates in 1-dim files**

While attempting to add two 1-dim files, DISCUS encountered different coordinates in the two input files. The adding is interrupted at this stage. The line number in which the error occurred, is displayed as well, check the input file for erroneous content.

**Error -26: Too many different atoms in crystal**

The maximum number of different atoms allowed in your implementation was exceeded. No more new types of atoms can be inserted into the structure. Check the chemistry of your crystal by the 'asym' and 'chem' commands. All atoms are considered different types that are chemically different, have different charge or a different temperature coefficient. If all types are needed, modify the parameter "maxscat" in the file "param.inc" and recompile the program. See chapter 9.1 of the manual for further information.

**Error -27: No atom of this type present in crystal**

An atom was selected for displacement by a wave or for plotting that does not exist within the crystal. Check the spelling of the atom name, and the chemistry of the crystal by the 'asym' and 'chem' commands.

**Error -28: Input parameters must be > zero**

This function/command requires non-negative parameters. Check the values of the parameters and the explanation for the function or command used for valid ranges of numerical input.

**Error -29: Too many different microdomain types**

The maximum number of microdomain types is defined by the parameter MAX\_MD\_TY in file "microdis.inc". If necessary adjust this parameter and recompile the program.

**Error -30: Unknown boundary type**

The boundary type of microdomains is not set to either of "block", "face", "fuzzy" or "sphere". Check the 'boundary' command at sublevel 'microdomains' for proper spelling.

**Error -31: Unknown distribution mode**

The distribution type for microdomains is not set to either of "lattice", "paracrystal" or "random". Check the 'distribution' command at sublevel 'microdomains' for proper spelling.

**Error -32: Length of vector is zero**

An attempt was made to calculate the angle between two vectors while one of them is of length zero. Check the parameters given on the 'bang' or 'rang' command for proper numbers.

**Error -33: No microdomain types defined yet**

An attempt was made to use the 'create' or 'run' command before any microdomains were defined by the 'insert' command. Check for proper sequence of commands and missing or misspelled 'insert' commands.

**Error -34: Form does not appear to be closed**

A microdomain type of boundary type 'face' was defined for which the form supplied by the 'bound' command does appear not to be closed. DISCUS checks whether the origin of the microdomain is inside the microdomain and the positions at  $1.0 \times 10^{-9}$  along each of the base vectors is outside the microdomain. If one of these conditions is not met, the error message is given. Check the 'boundary' commands used to define the microdomain type.

**Error -35: Volume of unit cell  $\leq$  zero**

The volume of the unit cell was calculated as zero or a negative value. Check the lattice parameters given in the input file. Are there any accidental "-" signs ? Do the angles form an impossible shape ?

**Error -36: Unsuitable file types for inverse Fourier**

Not every combination of file types is allowed for an inverse Fourier synthesis. Allowed file combinations are:

file a	file b
intensity	phase
amplitude	phase
real part	imaginary part

**Error -37: No filename defined yet**

An attempt was made to write output to or read from a file without defining a file name. DISCUS does not provide default names for the output of the 'plot' command or the input filenames. Check the 'outfile' command at sublevels 'output' and 'plot' or the 'content' command at sublevel 'microdomains'.

**Error -38: Unsuitable file types for Patterson**

Not every combination of file types is allowed for a Patterson synthesis. Allowed file combinations are:

file a	file b
intensity	any file type
amplitude	any file type
real part	imaginary part

**Error -39: All elements of correlation matrix zero**

Since the correlation matrix gives the probabilities for existence of a microdomain type, not all elements may be equal to zero. The default values at startup are 1 for the diagonal elements and zero for all other elements. display the correlation matrix with the command 'show corr' and check the 'ccol', 'cele' and 'crow' commands for proper values.



**Error -40: Microdomain type cannot be removed**

DISCUS can only remove those microdomain types from the list of types that have not yet been used by the 'create' command, i.e. only those microdomain types that have been inserted into the list by the 'insert' command after the last 'create' command.

**Error -41: File specifier must be "a" or "b"**

The two files for inverse Fourier and Patterson synthesis are referred to by "a" and "b". The first parameter on the 'file' and the 'type' commands must be either "a" or "b".

**Error -42: Type must be: inten,ampl,phase,real,imag**

The file type for inverse Fourier and Patterson synthesis must be specified as one of the following strings: "intensity", "amplitude", "phase", "real", "imaginary". They may be abbreviated to a unique string.

Check the spelling of the parameter.

**Error -43: Not enough parameter for filename format**

An attempt was made to generate a file name from a string like "text%dtext" without supplying enough numerical parameters. Check that the spelling of the string within " " is correct. Are there any unwanted %d combinations?. Check the number and type of parameters following the file name.

**Error -44: Right quotation mark missing in format**

An attempt was made to generate a file name from a string like "text%dtext" without supplying the right quotation mark. Check the line and provide the missing ".

**Error -45: Too many atoms in environment**

The indices of all atoms found are stored in the internal variable "res". More atoms were found that fit into the dimensions of "res". Restrict the search for the environment to a smaller shell or change the dimension of "res\_para" in file "param.inc".

**Error -46: Error reading title of structure**

An error occurred while reading the title line of a structure or unit cell file. Check the file for any garbage.

**Error -47: Error reading space group symbol**

An error occurred while reading the space group line of a structure or unit cell file. Check the file for any garbage.

**Error -48: Error reading lattice constants**

An error occurred while reading the lattice constants of a structure or unit cell file. Check the file for any garbage or accidental letters.

**Error -49: Error reading atom coordinates**

An error occurred while reading the atom coordinates of the atom listed. Check the file for any garbage. Is the line of the type Name x y z B Are there letters among the numerical values?

**Error -50: Wrong format for 1-dimensional data**

You tried to write a one-dimensional section of space as either Postscript, PGM or PPM Bitmap. One-dimensional data can only be written as "gnuplot" or "standard" format.

**Error -51: Unsuitable file type for SHELXL format**

If the file format is set to "shelxl" to calculate the inverse Fourier of Patterson form SHELXL data, the file type must be set to either of "fobs" or "fcalc". Check the 'type' command for erroneous input.

**Error -52: Unsuitable input value for SHELXL format**

If the file format is set to "shelxl" to calculate the inverse Fourier of Patterson form SHELXL data, the optional second parameter must be either of "fobs" or "fcalc". Check the input line for the 'form' command.

**Error -53: Too many different layer types**

You tried to define too many different layer types for a crystal with stacking faults. If necessary modify the value of "ST\_MAXTYPE" in file "stack.inc" and compile the program.

**Error -54: Index outside limits**

The value given is outside the proper limits allowed by this command. This usually means that an array element is outside the current dimension of an array, for example a correlation matrix or you are trying to include too many atoms in the crystal. Check the section on dimensions in the manual.

**Error -55: No layers created at all**

During the creation of the stacking faults, none of the layer types met the conditions imposed by the correlation matrix. Check the correlation matrix for proper values.

**Error -56: Av. Transl. in plane of moduli vectors**

The average translation vector, - calculated from the matrix of translation vectors and the correlation matrix or read via the 'aver' command is in the plane defined by the two moduli vector that were read using the 'modu' command. Use the 'show' command to check the values of the vectors. Correct the vectors by the 'aver', 'modu' commands.

**Error -57: Av. Translation is zero**

The average translation vector is a vector of length zero. Use the 'show' command to check the values of the vector. If the average vector is set explicitly by ==> 'set trans, fixed', then correct the vector by the 'aver' command. If the average vector is calculated from the translation and correlation matrix, the weighted sum of translation vectors is equal to zero. Use the 'show' command to check the values of the two matrices. Correct the values by the ==> 'trans', 'crow', 'ccol', 'cele' commands.

**Error -58: No atoms written to file**

No atoms were written to the plot file after the run command. Check the settings using the command 'show' and check the online help for further details. The most common problem is a too small value for 'thick'.

**Error -59: Invalid color or typ selected for atom**

The color or type selected for plotting atoms is invalid. Check the online help for a list of valid numbers.

**Error -60: Output value NOT allowed using lots**

You calculated the Fourier using ==> 'lots'. In this case you can write intensities only as output.

**Error -61: Too many additional generators**

You tried to specify too many additional generators through the "generator" key word in the structure file. The maximum number of generators is specified in the file "gen\_add.inc". If necessary, increase the value and compile DISCUS.

**Error -62: Too many additional symmetry operators**

You tried to specify too many additional symmetry operators through the "symmetry" key word in the structure file. The maximum number of symmetry operators is specified in the file "gen\_add.inc". If necessary, increase the value and compile DISCUS.

**Error -63: Molecule number outside limits**

You tried to select a molecule number that is less than zero or higher than the number of molecules in the crystal. Or the upper limit of a range of numbers could be less than the lower limit.

**Error -64: Molecule type outside limits**

You tried to select a molecule type that is less than zero or higher than the number of molecule types in the crystal. Or the upper limit of a range of numbers could be less than the lower limit.

**Error -65: Too many molecules created**

You created more molecules than fit into the corresponding arrays. Check the size of the crystal or the limits of loops used to create the molecules. If necessary adjust the maximum number of molecules in the file "molecule.inc" and compile DISCUS.

**Error -66: Too many molecules types created**

You created more molecule types than fit into the corresponding arrays. Check the number of molecules in the crystal file or the limits of loops used to create the molecules. If necessary adjust the maximum number of molecule types in the file "molecule.inc" and compile DISCUS.

**Error -67: Molecules have different number of atoms**

You tried to swap two molecules of different length. This mode is possible for molecules of equal length only.

**Error -68: Mode only available for molecules**

Rotational waves are can act only onto molecules, not onto individual atoms

**Error -69: Microdomains overlap**

An atom was found that is inside at least two microdomains. The atom number and the microdomain numbers are written onto the screen. Check the distribution of the microdomains and the corresponding microdomain radii. Either the two microdomains are too close to each other or the radii of the microdomains are too large.

**Error -70: Delta value must be in interval 0 -> 1**

A wave function of type "box" was chosen. The optional second parameter gives the width of the lower half of the box in multiples of the wave length. This value must be in the range 0 -> 1. The values 0 and 1 are not allowed either.

**Error -71: Too many points in direct space layer**

The number of data points you selected with the 'rhona' and 'rhono' commands exceeds the maximum number of data points DISCUS can handle. The maximum value is defined in the variable MAXQXY in the file "config.inc". If necessary adjust this parameter and compile DISCUS again.

**Error -72: Too many atom types in file**

The number of different atom types in the structure file is too large. Note that DISCUS treats atoms of the same name but with different B values as different atom types. The maximum value is defined in the variable MAXSCAT in 'config.inc'. If necessary adjust this parameter and compile DISCUS again.

**Error -73: Transform. requires too many generators**

The old symmetry operations of the space group are always changed into additional generators and the new space group is P1. This allows you to generate the structure from the list of atoms in the old asymmetric unit without running into a conflict between the symmetry and the unit cell shape. If you have transformed the old unit cell into an oddly shaped or into a very large new cell, the old translations will be transformed into a very large number of generators.

**Error -74: Molecule atom-number outside limits**

The argument to the variable "mol\_cont" is outside the number of atoms in this molecule, i.e. negative or too large. Check the number of atoms in this molecule through "mol\_cont[i1,0], where <i1> is the number of the molecule. If you are reading a crystal structure, then the number of atoms is too large to fit into the space allocated for molecules. Adjust the parameter MOLE\_MAX\_ATOM in the file "molecule.inc"

**Error -75:**

No bond valence parameters for atom pair

**Error -76: Function not available for molecules**

The selected items for the program ATOMS must be individual atoms, not molecules.

**Error -77: Unknown wave length symbol used**

You used a wave length symbol unknown to DISCUS. Check for a typing error and consult the list of allowed wave length symbols in the help file through "help wavelength". The wave length of nonlisted radiations must be explicitly given as real value in Angstroem.

**Error -78: Wrong optional parameter for HKLF4 format****Error -79: Too many atoms in result array**

Through finding environments or similar operations too many atoms were found. Reduce the radius of the environment that you are searching. Alternatively increase the size of the variable "MAXPAR\_RES" in file "param.inc"

**Error -80: Angle between Normal and Abszissa is 0**

The angle between the normal and the abszissa of the plot is zero degrees. Check the values you selected with the ==> 'uvw', 'hkl' and ==> 'abs' commands, and verify the angle in the ==> 'show' output.

**Error -81: Invalid flag for space type**

Commands like ==> 'projection' and the ==> 'function' "scalpro" take an optional flag that indicates whether the vectors are to be taken from direct or reciprocal space. Accordingly the flags are 2 to 4 characters long and may contain only the letters "d" and "r".

**Error -82: Invalid molecule character**

The character keyword for a molecule may only be "atom". If the molecule represents an extended object, used in small angle scattering calculations, it may be "cube", "cylinder", or "sphere". If the molecule represents a domain, it may be "cube", "cylinder", "sphere", or "fuzzy".

**Error -83: Molecule buildup failed**

DISCUS could not create the molecule. Check the molecule parameter lines in the structure file.

**Error -84: Error reading molecule parameters**

There seems to be a wrong keyword in the molecule entries.

**Error -85: Sharpened patternson requires HKLF4 file**

In order to calculate a sharpened Patterson function, the input file must be a SHELX HKLF4 file, i.e. contain h,k,l, I, sigma(i).

**Error -86: Unknown import format**

The import command was called with an unknown input format. Check the documentation on the ==> 'import' command for allowed values.

**Error -87: Different lattice constants**

You tried to include a domain into a host structure, whose lattice constants are different from those of the host crystal. Check the header of the domain file and the host structure for different metric. If necessary, ==> 'transform' the domain structure into the unit cell metric of the host crystal prior to the insertion.

**Error -88: Bravais types differ**

You tried to include a domain into a host structure, whose Bravais lattice type is different from that of the host crystal. Check the header of the domain file and the host structure for different space group symbols. If necessary, ==> 'transform' the domain structure into the unit cell metric of the host crystal prior to the insertion.

**Error -89: Unknown keyword in unit cell file**

There is an unknown keyword in the unit cell file. See help entry data for a complete listing of allowed keywords.

Check the spelling of all keywords in the file. Is the first keyword identical to "title" ? Did you include the keyword "atoms" as last keyword of the header? Do all molecules end with "molecule end" ?

**Error -90: Unknown diffractometer geometry**

The powder diffraction module can handle Lorentz/Polarisation corrections for the following geometries: bragg Bragg-Brentano Geometry neutron A neutron Diffractometer in Debye-Scherrer geometry synchrotron Synchrotron

**Error -91: Unexpected pseudoatom name read**

DISCUS was reading a structure file in which the atom names stand for a cluster. the cluster read, does not occur in the list of defines cluster names.

**Error -92: Error reading generators from structure**

DISCUS was reading a structure file which contained a "generator" key word when the error occurred.

Check that there are all 13 required parameters, that there are no alphabetic characters etc.

**Error -93: Unit cell constants <= zero**

At least one of the unit cell constants given in the current structure or unit cell file is zero or less than zero.

Check the 'cell' line in the input file for these errors.

**Error -94: Invalid domain descriptor in input file**

This error occurred while reading the file that contains descriptions of domains. The file name was set by ==> 'input' within the ==> 'domain' menu. The first parameter of the "domain" keywords in this file must be "character" or "content" or "file" or "fuzzy" or "end" or blank for the first "domain" keyword.

**Error -95: First domain keyword has parameters**

This error occurred while reading the file that contains descriptions of domains. The file name was set by ==> 'input' within the ==> 'domain' menu. The first "domain" keyword must not have any parameters, and must be terminated by a "domain end" keyword, once all settings have been defined.

**Error -96: Illegal keyword in domain input file**

This error occurred while reading the file that contains descriptions of domains. The file name was set by ==> 'input' within the ==> 'domain' menu.

After the standard header, a domain input file may contain only domain descriptors. These start with a "domain" keyword and end with a "domain end" keyword". Valid specifiers within a domain are "POST", "XAXI", "YAXI", "ZAXI", "CENT", "XDIM", "YDIM", "ZDIM"

**Error -97: Atom type outside proper limits**

The atom type number is less than zero, or larger than the current number of different atom types present in the crystal. Check the value, compare its value to n[2].

**Error -98: No atom types exist at present**

Wilson statistics requires that DISCUS knows which atom types were present in the sample, as a plot of  $\ln(\text{average}(\text{formfactor}^2)/\text{average}(\text{intensity}))$  versus  $\sin(\theta)/\lambda$  is evaluated. Insert the atom types into a (dummy) structure and make sure that their relative amount is close to the actual sample composition.

**Error -99: No wavelength has been set**

Wilson statistics requires that DISCUS knows which wave length was used in the experiment, as a plot of  $\ln(\text{average}(\text{formfactor}^2)/\text{average}(\text{intensity}))$  versus  $\sin(\theta)/\lambda$  is evaluated. Set the wave length through the ==> 'wvle' command in the ==> 'fourier' menu.

**Error -100: Space group symbol missing in cell file**

The required "spcgr" <symbol> line in the input file does not contain a space group symbol. Check the input file and provide a proper space group symbol.



**Error -101: Dimension of lots < 0 or > than crystal**

You are trying to use lots in the single crystal Fourier mode. At least one of the lots dimension is less than zero or larger than the crystal along that dimension. Check the line and inquire the crystal size with the ==> 'show cdim' command from the main menu.

**Error -102: Property value outside defined range**

While reading a cell file or structure, DISCUS encountered a property flag that is either zero or larger than currently defined. May be the input file was edited manually? Set the property flag to 1 to define the questionable atom as a standard atom type.

**Error -103: Too many atoms per unit cell**

The ==> 'aver' command in the ==> 'chem' menu experiences a situation in which there seem too many atom per unit cell. This error should never occur. Please document and report to the author.

**Error -104: Powder output not defined as TTH or Q**

You did a ==> 'powder' calculation without defining the axis as either ==> 'set axis,tth' or ==> 'set axis,q'. The output menu does not know which axis you intended. Please rerun the powder calculation with the correct axis set.

**Error -105: Atom number outside limits**

While reading an internally stored structure, the actual atom number is found to be larger than the maximum atom number stored in the header.

This error should never occur. Please document and report to the author.

**Error -106: HKL steps for complete powder must be >0**

All three step sizes for the HKL increment ==> 'set dh', 'set dk', 'set dl' are equal to zero! Make sure that at least one of these is set to a value larger than zero.

**Error -107: 2Theta limits or step width are illegal**

You are using the ==> 'set axis,tth' option and either: The upper limit of 2Theta is lesser than the lower limit, ==> 'set tthmin', ==> 'set tthmax' The 2Theta step size is zero or negative ==> 'set dtth'

Check the values and provide correct values.

**Error -108: Q limits or step width are illegal**

You are using the ==> 'set axis,q' option and either: The upper limit of Q is lesser than the lower limit, ==> 'set qmin', ==> 'set qmax' The Q step size is zero or negative ==> 'set dq'

Check the values and provide correct values.

**Error -109: Connectivity definition does not exist**

You are referencing a connectivity definition by name or number, that DISCUS cannot find. Was this connectivity never defined, or are you using a definition with a different atom type?

**Error -110: No connectivity definitions exist at all**

You are referencing a connectivity definition by name or number, but no definitions exist. Did you forget to define connectivity definitions or did you read a new unit cell/structure after the connectivity was defined?

**Error -111: Error reading SCAT instruction****Error -112:**

DISCUS encountered an error while reading the "scat" line in the current input file. Check if there are non printable characters in this line or if the line is empty. Error reading ADP instruction

DISCUS encountered an error while reading the "adp" line in the current input file. Check if there are non printable characters in this line or if the line is empty.

**Error -113: Could not find internal storage**

DISCUS cannot find the internally saved structure. Check for typing differences between the save and read lines.

**Error -114: Error allocating**

DISCUS could not allocate the memory space for internal storage.

Does your computer offer enough memory space, have you written huge files into the internal storage?

**Error -115: Different atom no on SCAT and ADP**

DISCUS uses the names and values on the "scat" and "adp" lines in the cell/structure input file to assign atom types. As there is a one to one correspondence between the parameters on these two lines, the number of parameters on both lines must be identical.

**Error -116: Could not find definition**

You referenced a connectivity definition which does not exist. Check that the connectivity definition was added and that the connectivities were run. Check for spelling errors on all respective commands

**Error -117: This DISCUS has NeXus support disabled**

In order for DISCUS to be able to write NeXus files, it must be compiled and linked with the NeXus library. Get the NeXus development library, with all dependencies and build a new DISCUS with NeXus.

**Error -118: No Fourier calculated yet, no output**

Before DISCUS can write an output file you need to calculate a diffraction pattern. This could be a: single crystal pattern via ==> 'fourier' powder pattern via ==> 'powder' single crystal pattern via stacking faults ==> 'stack' Patterson ==> 'patterson' inverse Fourier via ==> 'inverse'

**Error -119: Error reading atom number from RMCPROFILE**

DISCUS encountered an error while reading the number of atoms from an RMCPROFILE \*.cssr file. Check line 3 of the input file. DISCUS expects an integer number.

**Error -120: Conn. Name is equal to variable name**

You used a connectivity name that is already used as a variable name. This may lead to odd errors, as DISCUS would in most cases use the variable value instead of the connectivity name. You must choose a connectivity name that is not used as variable name.

**chem**

Errors at the chemistry level of DISCUS

**Error -1: Too many points for histogram**

The number of points you specified for the histogram exceeds the dimensions of the array. Check whether you need this many points. If so enlarge the value of "chem\_max\_bin" in the file "chem.inc"

**Error -2: Can't store all information in variable RES**

The number of items produced by the command exceeds the dimension of the result array.

**Error -3: Invalid range for bond-length histogram**

The minimum value of the bond lengths is +0.01, and the maximum value (parameter #2) must be greater than the minimum value. Check the parameters on the ==>'set bl' command.

**Error -4: Invalid SIGMA entered**

The sigmas must all be larger than zero. Check the parameters on the ==>'set fs', 'set wsig' commands.

**Error -5: Too many different atoms on ONE site**

The number of different atoms found on one site in the unit cell exceeds the dimensions of the array.

You have to set the value of "chem\_max\_ave" to a larger value and compile DISCUS again.

**Error -6: Atom name ALL not allowed for this command**

The ==>'corr' command requires explicit atom names. You cannot use the atom name "ALL".

**Error -7: Atoms need to be different**

The ==>'corr' command requires the two atom types to be different. The self correlation of atom1 with itself is stored by the 'freq' command as well. If you know that there are no other atom types in the given direction, set the second atom type to "void".

**Error -8: No neighbouring atoms found**

The 'freq' command found no neighbouring atoms for the given distance and direction. Check the online help for more information on 'neig'.

**Error -9: Invalid correlation vector index given**

The given correlation vector index was invalid. Check your input.

**Error -10: Invalid crystal site given**

A invalid site number for the current crystal was entered. Check that the structure is loaded and your site number is valid.

**Error -11: No correlations selected**

Before you can calculate correlations you need to define the calculation more using the 'set neig' command in the 'chem' sublevel.

**Error -12: Too many correlation definitions**

The command 'set corr,add' was executed but there is no more space for the additional correlation definition. Use 'set corr,rese' to discard previous definitions or adjust the value of CHEM\_MAX\_COR in 'chem.inc' and recompile DISCUS.

**Error -13: Correlation direction invalid**

The entered direction for the calculation of the displacement correlations is invalid (e.g. length zero), check your input.

**Error -14: Too many displacement directions**

You have entered too many displacement directions for the calculation of displacement correlations. Use 'set dir,rese' to discard the previous input our adjust the parameter CHEM\_MAX\_DIR in 'chem.inc' and recompile DISCUS.

**Error -15: No displacement directions selected**

You have tried to calculate displacement correlations but there are no directions defined that are used to obtain the displacement. Use the command 'set dir,..' to enter the desired values.

**Error -16: Failed to apply periodic boundaries**

DISCUS is not able to apply periodic crystal boundaries (should not happen).

**Error -17: Correlation fields require same mode**

To calculate correlation fields, all neighbour definitions must be made using the same mode, i.e. 'dist' or 'vec'. Redefine the neighbours to be used via the command 'set neig'. This condition is only valid for 2-dimensional correlation fields.

**Error -18: Correlation fields require same # vectors**

If working with neighbouring vector definitions to determine a 2D correlation field, the number of vectors must be the same for both directions.

**Error -19: No neighbouring molecules found**

For the current neighbour definitions no neighbouring molecules were found. Check your input.

**Error -20: No molecules present in crystal**

You cannot use the 'molecule' mode without having at least one molecule within the model crystal. Check the manual how to define molecules.

**Error -21: Molecule types need to be different**

In order to execute the selected operation (e.g. determine correlations), the selected molecule types need to be different. Check your input.

**Error -22: Command not available in molecule mode**

This particular command (e.g. aver) is not available in 'molecule' mode. Switch to 'atoms' mode using the command 'mode atom' and reenter the desired command.

**Error -23: Too many neighbouring atoms/molecules**

The current operation has found too many neighbouring atoms or molecules. Either restrict the neighbour definitions or adjust the variable CHEM\_MAX\_NEIG in the file 'config.inc' and recompile DISCUS.

**Error -24: Invalid correlation angle index given**

You tried to reference an environment of "angle" type that has not been defined. Check the list of defined environments and the values of parameters in the current command line.

**Error -25: Invalid range for bond-angle histogram**

The upper and lower limits of the bond angle histogram are invalid. Check that none is lower than 0 degrees or larger than 180 degrees, and that the upper limit is larger than the lower.

**Error -26: Invalid correlation environment index given**

You tried to reference an environment of "correlation" type that has not been defined. Check the list of defined environments and the values of parameters in the current command line.

**Error -27: No atoms present in crystal**

A number of commands can only be used if the crystal contains any atoms. Check the number of atoms "n[1]" present in your crystal.

**Error -28: Invalid correlation conn index given**

The given correlation conn index was invalid. Check your input. The index must be larger than 0.

**Error -29: Atom type outside valid range****four**

These are the possible error messages related to the calculation of the Fourier Transform

**Error -2: Invalid lot shape selected**

The only two supported lot shapes are 'eli' and 'box' ! Select a proper shape.

**Error -3: SIN(THETA)/LAMBDA > lookup table limits**

To speed up the calculation of the Fourier Transform, DISCUS uses a lookup table for SIN(THETA)/LAMBDA. You need to adjust the maximum h,k,l or the wave length. Alternatively, the values CFPKT and CFINC in the file 'config.inc' might be adjusted to the specific needs and DISCUS recompiled.

**Error -4: Component of increment vector is zero**

The corners of the plane in reciprocal space to be calculated are invalid since two or more corners are the same. Adjust your input.

**mmc**

These are the error messages of the MC section of DISCUS

**Error -1: Invalid or no energy type selected**

Before running a MC simulation, a valid energy definition (e.g. occupational correlations or distortions) must be selected using the 'set energy' command.

**Error -2: No valid move after 1000 cycles**

After 1000 cycles NO valid MC move was generated and DISCUS has terminated the simulation. Check your input.

**Error -3: Invalid mode selected for COCC MC run**

Occupational ordering requires the SWCHEM mode to be selected. Use the command 'set mode' to select the correct mode and restart the simulation.

**Error -4: Number of MC cycles is zero**

A Monte Carlo run was tried with zero number of refinement cycles. The number of cycles must be at least equal to one. Check the parameters on the last ==> 'set cycle' command, check that a ==> 'set cycle' command was used at all.

**Error -5: Number of feedback intervalls is zero**

The feedback intervall must be at least equal to one. Check the parameters on the last ==> 'set feed' command. check that a ==> 'set feed' command was used at all.

**pdf**

Errors at the PDF Level

**Error -1: Too many points in PDF**

The specified range and grid size for the PDF results in too many points. Either adjust your input or change the parameter MAXDAT in 'config.inc' and recompile DISCUS.

**Error -2: Cannot extend r-range for corr. convolution**

In order to carry out the convolution with the SINC function defined by the value of  $Q(\max)$ , the calculation range for the PDF needs to be extended. This error message occurs when there is not enough space for the additional data points. Either adjust your input or change the parameter MAXDAT in 'config.inc' and recompile DISCUS.

**Error -3: Crystal too large to create periodic bound.**

The size of the crystal is too large for the periodic boundary array DISCUS uses for rapid calculation of the PDF. Either use a smaller model crystal or change the parameter MAXBND in 'config.inc' and recompile DISCUS.

**Error -4: No structure defined yet ( $\geq 1$  atoms)**

You do not have a crystal structure with at least one atom. Create a structure first.

**Error -5: PDF data must start with  $r=Dr$** 

The data in the input file must be evenly spaced with step size  $\Rightarrow$  'set range'. The first data point must be at a distance value  $r$  equal to this set size.

Check the values on the  $\Rightarrow$  'set range' command and check the data file. If necessary repeat the determination of the experimental PDF or extrapolate the data from the current minimum  $r$ -value down to  $r(\min) = Dr$ .

**Error -6: PDF range fixed with data loaded****Error -7: Disable Gaussian mode and recalculate****Error -8: Atom type ALL not allowed**

The  $\Rightarrow$  'set partial' command must receive individual atom names, not the atom name "ALL".

**Error -9: No atoms in asymmetric unit**

The crystal does not contain any atoms in the asymmetric unit. It may have been read with the 'structure' command or generated via stacking faults. Under these circumstances you must use the exact mode in connection with the  $\Rightarrow$  'set bound'

**rmc**

Errors at the Reverse Monte Carlo Level



**Error -1: Too many experimental data planes**

This message appears if the user tries to read an additional plane of experimental data and the maximal number of planes is exceeded. If that many data planes are really needed, the variable RMC\_MAX\_PLANES has to be adjusted in the file 'config.inc' and DISCUS must be recompiled.

**Error -2: Too many experimental data points**

The experimental data file contains too many data points. In order to be able to use the particular data file the variable RMC\_MAX\_Q must be altered in 'config.inc' and DISCUS must be recompiled.

**Error -3: No atoms present in model crystal**

The command 'run' was entered but there is no model crystal present. The starting structure has to be set up before entering the RMC sublevel.

**Error -4: No experimental data present**

The command 'run' was entered but no experimental data are present. Use the command 'data' to read the experimental data.

**Error -5: Invalid method (x,n) selected**

When using the 'data' command an invalid scattering method was used. The present version of the program recognizes "neutron" and "x-ray" data. The method parameter can be abbreviated to a single letter. The input is not case sensitive.

**Error -7: Invalid plane selected**

An invalid data plane was selected. Use the command 'show' for a list of loaded data planes.

**Error -8: Invalid symmetry number selected**

The selected symmetry number is invalid. Use the command 'show' for a list of the current settings.

**Error -9: Invalid RMC mode selected**

An invalid RMC mode was selected. Check the online help entry for 'set mode' in the RMC sublevel for a list of allowed RMC moves.

**Error -10: No atom types selected for RMC run**

No atoms were selected for the RMC refinement. Use the command 'sele' to select the atom types to be used for the RMC run.

**Error -11: Only ONE atom type present in SWCHEM mode**

The RMC mode SWCHEM requires two different atom types to be selected. Vacancies are to be selected as VOID.

**Error -12: Displacements too small for SWDISP mode**

The displacements present in the starting structure are too small to use the SWDISP mode. The limit is set in 'rmc.f' if there is any need to change that.

**Error -13: Too many symmetrically equivalent planes**

The number of symmetrically equivalent planes exceeds the limit set by RMC\_MAX\_SYM in 'config.inc'. You can either adjust that value and recompile DISCUS or set the space group symbol in the structure file to P1 if you don't want to use the symmetry information.

**Error -14: No experimental data within given q limits**

There are no experimental data in the given q limits. Check the current settings with 'show' and adjust the limits using the command 'set range' in the RMC sublevel.

**Error -15: Invalid data type selected**

DISCUS can read PGM and NIPL files as input data. You selected a non supported data type. Check the online help for 'data' for a description of the valid data formats.

**Error -16: Invalid weighting scheme or weighting file**

The entered weighting scheme is invalid. Check the online help for 'data' for a description of the valid weighting schemes.

**Error -17: Data and weight file have different sizes**

If you read weights from a file, data file and weight file MUST have the same size.

**Error -18: Data file is not an ASCII PGM file**

In the present version, DISCUS can only read ASCII PGM files with the code P2 in the first line. Use the tool 'pnmnoraw' to transform from a binary to an ASCII version of your input file. The tool 'pnmnoraw' is part of the 'netpbm' package.

**Error -19: Invalid constrain entered**

The entered constrain is invalid. Check the online help for more information.

**Error -20: No valid move after 1000 display intervals**

After 1000 display intervals, DISCUS has not generated a single valid move. Check the settings for minimal allowed atom distances and all other RMC settings.

**Error -21: Number of LOTS exceeds maximum**

The number of 'lots' (crystal volumes to be averaged) exceeds the maximum number RMC\_MAX\_LOTS defined in 'config.inc'.

**A.60 space\_groups****List of allowed Space groups****a) Standard space groups settings**

# 1 : P1	# 2 : P-1		
# 3 : P2	# 4 : P21	# 5 : C2	# 6 : Pm
# 7 : Pc	# 8 : Cm	# 9 : Cc	# 10 : P2/m
# 11 : P21/m	# 12 : C2/m	# 13 : P2/c	# 14 : P21/c
# 15 : C2/c			
# 16 : P222	# 17 : P2221	# 18 : P21212	# 19 : P212121
# 20 : C2221	# 21 : C222	# 22 : F222	# 23 : I222
# 24 : I212121	# 25 : Pmm2	# 26 : Pmc21	# 27 : Pcc2
# 28 : Pma2	# 29 : Pca21	# 30 : Pnc2	# 31 : Pmn21
# 32 : Pba2	# 33 : Pna21	# 34 : Pnn2	# 35 : Cmm2
# 36 : Cmc21	# 37 : Ccc2	# 38 : Amm2	# 39 : Abm2
# 40 : Ama2	# 41 : Aba2	# 42 : Fmm2	# 43 : Fdd2
# 44 : Imm2	# 45 : Iba2	# 46 : Ima2	# 47 : Pmmm
# 48 : Pnnn	# 49 : Pccm	# 50 : Pban	# 51 : Pmma
# 52 : Pnna	# 53 : Pmna	# 54 : Pcca	# 55 : Pbam
# 56 : Pccn	# 57 : Pbcm	# 58 : Pnnm	# 59 : Pmmn
# 60 : Pbcn	# 61 : Pbca	# 62 : Pnma	# 63 : Cmcn
# 64 : Cmca	# 65 : Cmmm	# 66 : Cccm	# 67 : Cmma
# 68 : Ccca	# 69 : Fmmm	# 70 : Fddd	# 71 : Immm
# 72 : Ibam	# 73 : Ibca	# 74 : Imma	
# 75 : P4	# 76 : P41	# 77 : P42	# 78 : P43
# 79 : I4	# 80 : I41	# 81 : P-4	# 82 : I-4
# 83 : P4/m	# 84 : P42/m	# 85 : P4/n	# 86 : P42/n
# 87 : I4/m	# 88 : I41/a	# 89 : P422	# 90 : P4212
# 91 : P4122	# 92 : P41212	# 93 : P4222	# 94 : P42212
# 95 : P4322	# 96 : P43212	# 97 : I422	# 98 : I4122
# 99 : P4mm	# 100 : P4bm	# 101 : P42cm	# 102 : P42nm
# 103 : P4cc	# 104 : P4nc	# 105 : P42mc	# 106 : P42bc
# 107 : I4mm	# 108 : I4cm	# 109 : I41md	# 110 : I41cd
# 111 : P-42m	# 112 : P-42c	# 113 : P-421m	# 114 : P-421c
# 115 : P-4m2	# 116 : P-4c2	# 117 : P-4b2	# 118 : P-4n2
# 119 : I-4m2	# 120 : I-4c2	# 121 : I-42m	# 122 : I-42d
# 123 : P4/nmm	# 124 : P4/mcc	# 125 : P4/nbm	# 126 : P4/nnc
# 127 : P4/mbm	# 128 : P4/mnc	# 129 : P4/nmm	# 130 : P4/ncc
# 131 : P42/nmc	# 132 : P42/mcm	# 133 : P42/nbc	# 134 : P42/nnn
# 135 : P42/mbc	# 136 : P42/mnm	# 137 : P42/nmc	# 138 : P42/ncc
# 139 : I4/nmm	# 140 : I4/mcm	# 141 : I41/amd	# 142 : I41/acd
# 143 : P3			
# 144 : P31		# 145 : P32	
# 146 : R3: hexagonal axes		# 147 : P-3	
# 148 : R-3: hexagonal axes		# 149 : P312	
# 150 : P321		# 151 : P3112	
# 152 : P3121		# 153 : P3212	

```

# 154 : P3221
# 156 : P3m1
# 158 : P3c1
# 160 : R3m: hexagonal axes
# 162 : P-31m
# 164 : P-3m1
# 166 : R-3m: hexagonal axes
# 168 : P6
# 172 : P64
# 176 : P63/m
# 180 : P6222
# 184 : P6cc
# 188 : P-6c2
# 192 : P6/mcc
# 195 : P23
# 199 : I213
# 203 : Fd-3
# 207 : P432
# 211 : I432
# 215 : P-43m
# 219 : F-43c
# 223 : Pm-3n
# 227 : Fd-3m

# 155 : R32: hexagonal axes
# 157 : P31m
# 159 : P31c
# 161 : R3c: hexagonal axes
# 163 : P-31c
# 165 : P-3c1
# 167 : R-3c: hexagonal axes
# 170 : P65
# 174 : P-6
# 178 : P6122
# 182 : P6322
# 186 : P63mc
# 190 : P-62c
# 194 : P63/mmc
# 197 : I23
# 201 : Pn-3
# 205 : Pa-3
# 209 : F432
# 213 : P4132
# 217 : I-43m
# 221 : Pm-3m
# 225 : Fm-3m
# 229 : Im-3m

# 171 : P62
# 175 : P6/m
# 179 : P6522
# 183 : P6mm
# 187 : P-6m2
# 191 : P6/mmm
# 198 : P213
# 202 : Fm-3
# 206 : Ia-3
# 210 : F4132
# 214 : I4132
# 218 : P-43n
# 222 : Pn-3n
# 226 : Fm-3c
# 230 : Ia-3d

```

## b) non standard settings of space groups

```

# 231 : P112:      # 3: P2 : unique c axis
# 232 : P1121:     # 4: P21 : unique c axis
# 233 : A121:      # 5: C2 : unique b axis: cell choice 2
# 234 : I121:      # 5: C2 : unique b axis: cell choice 3
# 235 : A112:      # 5: C2 : unique c axis: cell choice 1
# 236 : B112:      # 5: C2 : unique c axis: cell choice 2
# 237 : I112:      # 5: C2 : unique c axis: cell choice 3
# 238 : P11m:      # 6: Pm : unique c axis
# 239 : P1n1:      # 7: Pc : unique b axis: cell choice 2
# 240 : P1a1:      # 7: Pc : unique b axis: cell choice 3
# 241 : P11a:      # 7: Pc : unique c axis: cell choice 1
# 242 : P11n:      # 7: Pc : unique c axis: cell choice 2
# 243 : P11b:      # 7: Pc : unique c axis: cell choice 3
# 244 : A1m1:      # 8: Cm : unique b axis: cell choice 2
# 245 : I1m1:      # 8: Cm : unique b axis: cell choice 3
# 246 : A11m:      # 8: Cm : unique c axis: cell choice 1
# 247 : B11m:      # 8: Cm : unique c axis: cell choice 2
# 248 : I11m:      # 8: Cm : unique c axis: cell choice 3
# 249 : A1n1:      # 9: Cc : unique b axis: cell choice 2
# 250 : I1a1:      # 9: Cc : unique b axis: cell choice 3
# 251 : A11a:      # 9: Cc : unique c axis: cell choice 1
# 252 : B11n:      # 9: Cc : unique c axis: cell choice 2
# 253 : I11b:      # 9: Cc : unique c axis: cell choice 3
# 254 : P112/m:    # 10: P2/m : unique c axis
# 255 : P1121/m:   # 11: P21/m : unique c axis
# 256 : A12/m1:    # 12: C2/m : unique b axis: cell choice 2
# 257 : I12/m1:    # 12: C2/m : unique b axis: cell choice 3
# 258 : A112/m:    # 12: C2/m : unique c axis: cell choice 1
# 259 : B112/m:    # 12: C2/m : unique c axis: cell choice 2
# 260 : I112/m:    # 12: C2/m : unique c axis: cell choice 3
# 261 : P12/n1:    # 13: P2/c : unique b axis: cell choice 2
# 262 : P12/a1:    # 13: P2/c : unique b axis: cell choice 3
# 263 : P112/a:    # 13: P2/c : unique c axis: cell choice 1
# 264 : P112/n:    # 13: P2/c : unique c axis: cell choice 2
# 265 : P112/b:    # 13: P2/c : unique c axis: cell choice 3
# 266 : P121/n1:   # 14: P21/c : unique b axis: cell choice 2
# 267 : P121/a1:   # 14: P21/c : unique b axis: cell choice 3
# 268 : P1121/a:   # 14: P21/c : unique c axis: cell choice 1

```

```

# 269 : P1121/n:      # 14: P21/c : unique c axis: cell choice 2
# 270 : P1121/b:      # 14: P21/c : unique c axis: cell choice 3
# 271 : A12/n1:        # 15: C2/c : unique b axis: cell choice 2
# 272 : I12/a1:        # 15: C2/c : unique b axis: cell choice 3
# 273 : A112/a:        # 15: C2/c : unique c axis: cell choice 1
# 274 : B112/n:        # 15: C2/c : unique c axis: cell choice 2
# 275 : I112/b:        # 15: C2/c : unique c axis: cell choice 3
# 276 : Pnnn:          # 48: origin choice 2
# 277 : Pban:          # 50: origin choice 2
# 278 : Pmmn:          # 59: origin choice 2
# 279 : Ccca:          # 68: origin choice 2
# 280 : Fddd:          # 70: origin choice 2
# 281 : P4/n:          # 85: origin choice 2
# 282 : P42/n:         # 86: origin choice 2
# 283 : I41/a:         # 88: origin choice 2
# 284 : P4/nbm:        # 125: origin choice 2
# 285 : P4/nnc:        # 126: origin choice 2
# 286 : P4/nmm:        # 129: origin choice 2
# 287 : P4/ncc:        # 130: origin choice 2
# 288 : P42/nbc:       # 133: origin choice 2
# 289 : P42/nnn:       # 134: origin choice 2
# 290 : P42/nmc:       # 137: origin choice 2
# 291 : P42/ncm:       # 138: origin choice 2
# 292 : I41/amd:       # 141: origin choice 2
# 293 : I41/acd:       # 142: origin choice 2
# 294 : R3:            # 146 rhombohedral axes
# 295 : R-3:           # 148 rhombohedral axes
# 296 : R32:           # 155 rhombohedral axes
# 297 : R3m:           # 160 rhombohedral axes
# 298 : R3c:           # 161 rhombohedral axes
# 299 : R-3m:          # 166 rhombohedral axes
# 300 : R-3c:          # 167 rhombohedral axes
# 301 : Pn-3:          # 201: origin choice 2
# 302 : Fd-3:          # 203: origin choice 2
# 303 : Pn-3n:         # 222: origin choice 2
# 304 : Pn-3m:         # 224: origin choice 2
# 305 : Fd-3m:         # 227: origin choice 2
# 306 : Fd-3c:         # 228: origin choice 2
# 307 : Pbnm:          # 62: Pnma non-standard orientation
# 308 : Pmnn:          # 58: Pnm non-standard orientation

```

## A.61 wavelength

### List of allowed wavelength symbols

TIA1	2.748410	TIA2	2.752070	TIB	2.513810	TI	2.740000
CRA1	2.28970	CRA2	2.29361	CRB	2.0848	CR	2.2909
FEA1	1.193604	FEA2	1.93998	FEB	1.75653	FE	1.9373
COA1	1.788920	COA1	1.792780	COB	1.620750	CO	1.780000
CUA1	1.54056	CUA2	1.54439	CUB	1.39217	CU	1.5418
MOA1	0.709309	MOA2	0.71335	MOB	0.63225	MO	0.7107
AGA1	0.55941	AGA2	0.56380	AGB	0.49701	AG	0.5608
TAA1	0.215484	TAA2	0.220290	TAB	0.190076	TA	0.560800
WA1	0.208992	WA2	0.213813	WB	0.184363,	W	0.560800
AUA1	0.180185	AUA2	0.185064	AUB	0.158971,	AU	0.560800

## A.62 atom\_names

### List of allowed atom names

```

H      H.    H1-  D.    HE    LI    LI1  BE    BE2+ B    C    N    O    O1-
O2-    F    F1-  NE    NA    NA1+ MG    MG2+ AL    AL3+ SI    SI.  SI4+ S
P      CL    CL1- AR    K      K1+  CA    CA2+ SC    SC3+ TI    TI2+ TI3+ TI4+
V      V2+   V3+   V5+   CR    CR2+ CR3+ MN    MN2+ MN3+ MN4+ FE    FE+  FE3+
CO     CO2+  CO3+  NI     NI2+  NI3+  CU    CU1+  CU2+  ZN    ZN2+  GA    GA3+  GE
GE4+   AS    SE    BR    BR1-  KR    RB    RB1+  SR    SR2+  Y     Y3+   ZR    ZR4+
NB     NB3+  NB5+  MO     MO3+  MO5+  MO6+  TC    RU    RU3+  RU+   RH    RH3+  RH4+
PD     PD2+  PD4+  AG     AG1+  AG2+  CD    CD2+  IN    IN3+  SN    SN2+  SN4+  SB
SB3+   SB5+  TE    I      I1-   XE    CS    CS1+  BA    BA2+  LA    LA3+  CE    CE3+
CE4+   PR    PR3+  PR4+  ND     ND3+  PM    PM3+  SM    SM3+  EU    EU2+  EU3+  GD
GD3+   TB    TB3+  DY     DY3+  HO     HO3+  ER    ER3+  TM    TM3+  YB    YB2+  YB3+
LU     LU3+  HF     HF4+  TA     TA5+  W      W6+   RE    OS    OS4+  IR    IR3+  IR4+
PT     PT2+  PT4+  AU     AU1+  AU3+  HG     HG1+  HG2+  TL    TL1+  TL3+  PB    PB2+
PB4+   BI    BI3+  BI5+  PO     AT    RN    FR    RA    RA2+  AC    AC3+  THE   TH4+
PA     U      U3+   U4+   U6+   NP     NP3+  NP4+  NP6+  PU     PU3+  PU4+  PU6+  AM
CM     BK    CF

```

The symbols ending with a period have special meanings as listed:

```

H.    Hydrogen HF scattering factor
C.    Carbon HF scattering factor
SI.   Silicon HF scattering factor
D.    Deuterium (for neutron diffraction only)
O2-   Oxygen, taken from Acta Cryst. 19, p 486 (1965).

```

Other special names:

```

E1-   Single electron with form factor =1 independent of sin( )/

```

Other names may be used as well, if scattering curves are supplied for these names. See the 'delf, and 'scat' commands at 'fourier' sublevel for details.

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