# **Electron diffraction CIF dictionary**

#### CIF ED HEAD

This category is the Head category for an extension dictionary used for crystallographic structure determination using electron diffraction.

#### **CHEMICAL**

#### chemical.absolute configuration

(Text)

\_chemical\_absolute\_configuration

Necessary conditions for this assignment are given by Flack, H. D. & Bernardinelli, G. (1999). *Acta Cryst.* A**55**, 908–915. Flack, H. D. & Bernardinelli, G. (2000). *J. Appl. Cryst.* **33**, 1143–1148.

The data value must be one of the following:

ne aata tarae	must be one of the following.
rm	'reference molecule' Absolute configuration established by the structure determination of a compound containing a chiral reference molecule of known absolute configuration.
ad	'anomalous dispersion' Absolute configuration established by a-d effects in diffraction measurements on the crystal.
dyn	Absolute configuration determined directly from dynamical refinement.
rmad	'rm + ad' Absolute configuration established by the structure determination of a compound containing a chiral reference molecule of known absolute configuration and confirmed by a-d effects in diffraction measurements on the crystal.
syn	'synthetic' Absolute configuration has not been established by anomalous-dispersion effects in diffraction measurements on the crystal. The enantiomer has been assigned by reference to an unchanging chiral centre in the synthetic procedure.
unk	'unknown' No firm chemical or a-d evidence for an assignment is available. An arbitrary choice of enantiomer has been made.
	Inapplicable.

# COMPUTING

## \_computing.sample\_tracking

(Text)

Reference to software used to track the position of a sample with respect to the diffraction source beam. Typically used in 3D electron diffraction. More details of the tracking method may be provided in \_diffrn\_measurement.sample\_tracking\_method.

Example: 'PyFast-ADT (Santucci, 2024)'

## **DIFFRN**

## \_diffrn.precession\_angle

(Real;

Angle at the sample of tilt from the normal direction of a radiation beam moved in a precessional scan.

The permitted range is  $0.0 \to \infty$ .

## $\_\mathtt{diffrn.precession}\_\mathtt{angle}\_\mathtt{su}$

(Real; °)

Standard uncertainty of \_diffrn.precession\_angle.

#### DIFFRN\_DETECTOR

# $\_$ diffrn $\_$ detector.ed $\_$ calibration $\_$ constant (Real; $\mathring{A}^{-1}$ per pixel)

The electron diffraction camera constant calibration factor.

Reference: Millette, J. (1987). The Microscope, 35, 107–117.

The permitted range is  $0.0 \rightarrow \infty$ .

#### DIFFRN\_MEASUREMENT

## \_diffrn\_measurement.integration

(Code)

Code describing the technique used to sample as completely as possible the accessible range of reciprocal space.

The data value must be one of the following:

beam-tilt	The radiation beam is tilted in order to collect intensities in dif-
	ferent regions of reciprocal space while the sample remains
	stationary.
continuous	Synonym for 'rotation' commonly used in 3D electron diffrac-
	tion.
rotation	The detector records diffracted intensities continuously while
	the sample is rotated. This is the usual mode for X-ray crys-
	tallography.
stepwise	The detector records diffracted intensities at intervals while
	the sample is panned or tilted in discrete steps. Often used in
	3D electron diffraction.

#### \_diffrn\_measurement.method\_precession (Code)

Yes or no flag indicating if the radiation beam illuminates the sample at an angle that is rotated about a precession axis.

The data value must be one of the following:

yes	Precession method is used.
У	Abbreviation for 'yes'.
no	Precession method is not used.
n	Abbreviation for 'no'.

#### \_diffrn\_measurement.sample\_tracking (Code)

Yes or no flag indicating if a tracking method to follow the crystal was used in the data acquisition. Typically used to track very small crystals in electron diffraction experiments.

The data value must be one of the following:

yes	A crystal tracking method is used.
У	Abbreviation for 'yes'.
no	A crystal tracking method is not used.
n	Abbreviation for 'no'.

#### \_diffrn\_measurement.sample\_tracking\_method

(Text)

Description of the method used to follow the crystal during data collection.

Example: 'Electron-beam tracking by PyFast-ADT'

## **DIFFRN\_RADIATION**

## \_diffrn\_radiation.illumination\_mode (C

Code describing the optical configuration of the incident beam at the sample. For electron diffraction, parallel illumination is the normal mode in transmission electron microscopy (TEM), convergent-beam in scanning transmission electron microscopy (STEM).

The data value must be one of the following:

convergent Convergent-beam illumination
parallel Parallel illumination

#### DIFFRN\_SOURCE

## \_diffrn\_source.convergence\_angle

(Real: °)

The angle of convergence of the source beam (for example in convergent-beam electron diffraction). The convergence angle is the angular spread of the cone of illumination (i.e.  $2\alpha$ , where the semi-angle  $\alpha$  is measured from the axis defining the beam direction).

The permitted range is  $0.0 \to \infty$ .

## \_diffrn\_source.ed\_probe\_formation (Text)

Enumerated code for the technique in electron diffraction used to determine the probe size to illuminate the appropriate region of the sample by the incident radiation beam. (Terminology from https://www.globalsino.com/EM/page2695.html.)

The data value must be one of the following:

 $\begin{array}{ll} \text{microdiffraction} & Probe \ size \ 500\text{--}100 \ nm \\ \text{nanodiffraction} & Probe \ size \ <100 \ nm \end{array}$ 

SAED Selected-area electron diffraction.

#### \_diffrn\_source.size

(Text)

\_diffrn\_source\_size

Description of the collimated source beam as viewed from the sample.

Examples: '8mm x 0.4 mm fine-focus', 'broad focus',

'300 nm electron beam in NBED (nano-beam electron diffraction)',

'1 micrometre aperture size in SAED (selected-aperture el. diffr.)'

#### EXPTL\_CRYSTAL

#### \_exptl\_crystal.mosaic\_block\_size (Real; Å)

Isotropic and resolution-independent term representing the average size of mosaic domains in the crystal specified in ångströms. Larger size indicates better ordered crystals.

The permitted range is  $0. \to \infty$ .

#### \_exptl\_crystal.mosaic\_block\_size\_su (Real; Å)

Standard uncertainty of \_exptl\_crystal.mosaic\_block\_size.

#### \_exptl\_crystal.mosaic\_method (Text)

How parameters derived from the spot shape (such as mosaic block size and rotation, beam divergence, and crossfire) and their errors were estimated. This can be a written description or a citation to a specific software package that determined these parameters.

Example: 'Determined from rocking curve.'

## \_exptl\_crystal.mosaicity (Real; °

Isotropic approximation of the distribution of mis-orientation angles specified in degrees of all the mosaic domain blocks in the crystal, represented as a standard deviation. Here, a mosaic block is a set of contiguous unit cells assumed to be perfectly aligned. Lower mosaicity indicates better ordered crystals. See for example: Nave, C. (1998). *Acta Cryst.* D**54**, 848–853. Note that many software packages estimate the mosaic rotation distribution differently and may combine several physical properties of the experiment into a single mosaic term. This term will help fit the modelled spots to the observed spots without necessarily being directly related to the physics of the crystal itself.

The permitted range is  $0. \to \infty$ .

#### $\_\mathtt{exptl\_crystal.mosaicity\_su}$ (Real; $^{\circ}$ )

Standard uncertainty of \_exptl\_crystal.mosaicity.

#### REFINE

## \_refine.diffraction\_theory

(Code)

A code describing the approach to the calculation of the diffracted intensities. In the kinematical approximation, interactions of diffracted waves with matter are neglected, and integrated intensities of the Bragg reflections are proportional to the square of the modulus of the structure factor. In the dynamical calculation, interactions between diffracted waves are taken into account, and the proportionality of the intensities and the structure factors is not preserved.

The data value must be one of the following:

dynamical Interactions bet

Interactions between diffracted waves and matter are taken into account when calculating the amplitudes of the

diffracted beams.

kinematical Interactions between diffracted waves and matter are ignored when calculating the amplitudes of diffracted beams.

#### REFINE\_DIFF

## \_refine\_diff.density\_max

 $(Real, e Å^{-1})$ 

\_refine\_diff\_density\_max

\_refine.diff\_density\_max

Maximum density value in a difference Fourier map.

The permitted range is  $-100. \rightarrow \infty$ .

#### \_refine\_diff.density\_max\_su

(Real, e  $\text{Å}^{-1}$ )

\_refine\_diff\_density\_max\_su

\_refine.diff\_density\_max\_esd

Standard uncertainty of \_refine\_diff.density\_max.

# \_refine\_diff.density\_min

 $(Real, e Å^{-1})$ 

\_refine\_diff\_density\_min

\_refine.diff\_density\_min

Minimum density value in a difference Fourier map.

The permitted range is  $-\infty \to 100$ ..

## \_refine\_diff.density\_min\_su

 $(Real, e Å^{-1})$ 

\_refine\_diff\_density\_min\_su

\_refine.diff\_density\_min\_esd

Standard uncertainty of refine diff.density min.

## \_refine\_diff.density\_RMS

(Real, e  $\text{Å}^{-1}$ )

 $\_refine\_diff\_density\_RMS$ 

\_refine.diff\_density\_RMS

Root mean square density value in a difference Fourier map. This value is measured with respect to the arithmetic mean density and is derived from summations over each grid point in the asymmetric unit of the cell. This quantity is useful for assessing the significance of \*\_min and \*\_max values, and also for defining suitable contour levels.

The permitted range is  $-100. \rightarrow 100.$ 

#### \_refine\_diff.density\_RMS\_su

(Real, e  $Å^{-1}$ )

\_refine\_diff\_density\_RMS\_su

\_refine.diff\_density\_RMS\_esd

Standard uncertainty of \_refine\_diff.density\_RMS.

#### REFINE\_LS

# \_refine\_ls.abs\_structure\_z-score

(Real)

The z-score is a measure of confidence in an absolute structure determination as described by Klar *et al.* (2023), based on the method of Le Page *et al.* (1990). For centrosymmetric structures, the only permitted value, if the data item is present, is 'inapplicable', represented by '.'

References: Klar, P. B. *et al.* (2023). *Nature Chem.* **15**, 848–855. Le Page, Y., Gabe, E. J. & Gainsford, G. J. (1990). *J. Appl. Cryst.* **23**, 406–411.

#### \_refine\_ls.sample\_shape\_details

(Text)

Additional details describing the sample thickness when calculated during dynamical refinement where this cannot be defined completely by the formula given in \_refine\_ls.sample\_shape\_expression.

Reference: Palatinus, L., Petricek, V. & Correa, C. A. (2015). *Acta Cryst.* A**71**, 235–244.

## \_refine\_ls.sample\_shape\_expression

(Text)

Formula for the sample thickness distribution when the sample thickness is considered during dynamical refinement.

Reference: Palatinus, L., Petricek, V. & Correa, C. A. (2015). *Acta Cryst.* A**71**, 235–244.

 $F(\t) = 1 - (1 - \t^2)^1/2$ 

; (The cumulative distribution function in terms of the reduced thickness  $\tau$  of a sample treated as a cylinder.  $\tau = t/t_m$ , where t is the thickness at the point illuminated and  $t_m$  the maximum thickness of the cylinder (*i.e.* its diameter).)

# \_refine\_ls.sample\_thickness

(Real; nanometres)

The refined sample thickness as determined by dynamical refinement

Reference: Palatinus, L., Petricek, V. & Correa, C. A. (2015). *Acta Cryst.* A**71**, 235–244.

The permitted range is  $0. \to \infty$ .

Example: '35 (2)'

# \_refine\_ls.sample\_thickness\_su

(Real; nanometres)

Standard uncertainty of \_refine\_ls.sample\_thickness.