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Insights into lattice oxygen and strains of oxide-derived copper for ammonia electrosynthesis from nitrate

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Electrocatalytic NO₃⁻ reduction (eNO3RR) is a sustainable method for purification of NO₃⁻ wastewater and NH₃ recovery. Cu-based catalysts are promising for eNO3RR, but insufficient active hydrogen (*H) supply and *NO₂ poison of active sites have hindered their performance, and the catalytic mechanism remains ambiguous. Here, we report oxide-derived copper nanosheet arrays (OD-Cu NSs) with residual lattice oxygen and lattice strains to enhance NH₃ synthesis from eNO3RR. It is efficient for NH₃ synthesis with high Faradaic efficiencies of 88.7-99.7% and maximum NH₃ yield of 6.20 mmol·h⁻¹·cm⁻² at neutral solution, 10-140 mM NO₃⁻ and 50-1500 mA·cm⁻². Experimental and theoretical results reveal that lattice oxygen regulates the electronic structure of OD-Cu NSs and promotes *NO₂ conversion, while lattice strain enhances *H generation from water dissociation, resulting in the good performance for NH₃ synthesis. The applicability of OD-Cu NSs is proved by the high recovery of ammonia compound from eNO3RR.

NH $_3$ is an important chemical with extensive applications in agriculture, chemical industries, and the energy sectors 1 . However, NH $_3$ synthesis currently relies on the conventional Haber-Bosch process 2 . It requires high temperature and pressure, which consumes about 1-2% of global energy and emits large amounts of ${\rm CO}_2{}^3$. Consequently, a green and low-energy method for NH $_3$ synthesis is urgently needed. Electrocatalytic NO $_3{}^-$ reduction (eNO3RR) has emerged as a promising alternative for NH $_3$ synthesis owing to its mild operating conditions and low energy consumption (<0.5 kWh·g $_{\rm N}{}^{-1}$) 4 . Moreover, eNO3RR can tackle NO $_3{}^-$ pollution in water, one of the major environmental problems resulted from agricultural and industrial activities. It not only purifies water but also recoveries NH $_3$ from NO $_3{}^-$ rich wastewater.

NH₃ synthesis from eNO3RR is highly dependent on electrocatalysts. Among the various catalysts reported, Cu-based catalysts are good candidates as the high electron density of Cu d-orbital is favorable for NO₃⁻ adsorption and the initiation of eNO3RR⁵. However, eNO3RR process involves complex steps, including proton-coupled electron transfer, the adsorption and hydrogenation of reactants and intermediates⁶. The eNO3RR on Cu-based catalysts is hindered by the inherent defects of *NO₂ poison (intermediates for eNO3RR) and weak *H supply on Cu^{7,8}. The Cu surface has abundant free electrons. It forms strong adsorption with *NO₂ through the electron coupling effect⁹, poisoning the active sites¹⁰. Moreover, the hydrogenation of intermediates is also crucial for eNO3RR. Cu is known to be inefficient for *H generation from water dissociation due to its high energy barrier¹¹. The slow *H supply limits the hydrogenation kinetics for eNO3RR. To enhance NH₃ synthesis from eNO3RR, *NO₂ conversion or *H supply has been regulated by the strategies such as metal/non-metal atom doping^{7,12,13}, microstructure regulation¹⁴, design of atomically dispersed Cu^{15,16} and Cu/Cu₂O heterojunction^{17,18}. Despite the progresses have been made, simultaneous optimization of *NO2 conversion and *H supply to enhance NH₃ electrosynthesis from eNO3RR has been rarely reported. Meanwhile, tailoring active sites that are both

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Oxide-derived copper (OD-Cu) is a promising catalyst for electrocatalytic reduction¹⁹. Given the strong electron-withdrawing property of oxygen, residual lattice oxygen serves as a negatively charged center in the Cu lattice, which can limit the flow of free electrons. probably weakening *NO₂ adsorption. However, previous studies have focused on the crystalline phases and facets of OD-Cu catalysts, overlooking the impact of residual lattice oxygen in eNO3RR. Moreover, OD-Cu has abundant lattice strains due to the oxygen atoms separating out from Cu oxide²⁰. The lattice strains can modulate the activity and product selectivity of electrocatalysts by breaking the linear scaling relationship²¹. Li et al. observed lattice strains in OD-Cu, but their effect on catalysis was not investigated²². It was found later that lattice tensile strains on Cu could lead to hydrogen evolution²³. However, the role of lattice tensile strain in *H supply from water splitting hasn't been reported. Meanwhile, the catalytic mechanism of OD-Cu remains insufficiently understood. For example, in situ generated Cu/Cu₂O interface, pure Cu or Cu₂O phase have been reported as active sites for eNO3RR²⁴⁻²⁷. The precise active sites are still under debate.

In this work, oxide-derived copper nanosheet arrays (OD-Cu NSs) with residual lattice oxygen and lattice strains were designed to regulate $*NO_2$ adsorption strength and *H supply during eNO3RR. The experimental and density functional theory (DFT) results revealed that

residual lattice oxygen modulated the electronic state of the Cu surface, reducing *NO $_2$ adsorption and facilitating its conversion. The lattice tensile strains enhanced *H supply from water dissociation. The resulted OD-Cu NSs achieved high Faradaic efficiencies (FEs) of 88.7–99.7% for NH $_3$ production at neutral solution and 50–1500 mA·cm $^-$ 2, surpassing the recently reported catalysts. Its potential application has been proved by the efficient recovery of NH $_3$ and NH $_4$ Cl from NO $_3$ $^-$ reduction.

Results

Synthesis and characterization of OD-Cu NSs

OD-Cu NSs were synthesized on copper foam via chemical oxidation and electrochemical reduction²⁸ (Fig. 1a). The CuO nanosheets were grown on copper foam by oxidation in sodium hydroxide and ammonium persulfate solution. OD-Cu NSs were obtained by electrochemical reduction of CuO nanosheets, in which the reduction potential was determined from its cyclic voltammogram (Supplementary Fig. 1). As a comparison, copper nanosheet arrays (Cu NSs) were prepared by calcination of CuO nanosheets under H₂. The calcination temperature was selected based on deep removal of lattice oxygen through H₂-temperature programmed reduction (H₂-TPR) test (Supplementary Fig. 2). Scanning electron microscopy (SEM) shows the ultrathin and smooth nanosheet morphology of OD-Cu NSs (Fig. 1b and Supplementary Fig. 3). However, the nanosheet surfaces of Cu NSs are rough due to the migration and agglomeration of surface atoms

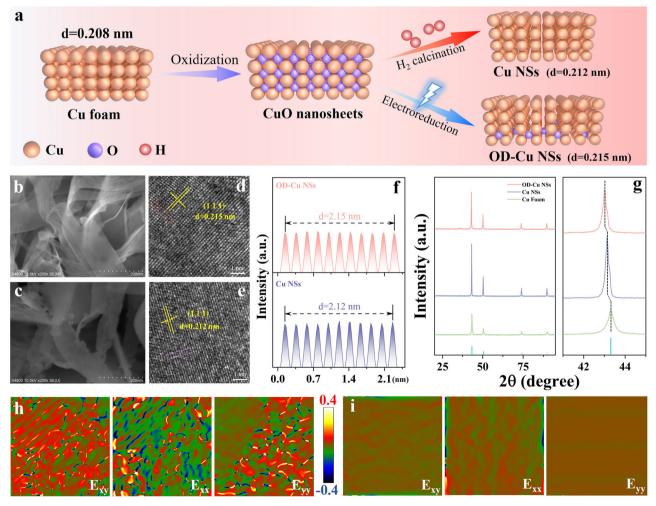


Fig. 1 | **Synthetic scheme and morphological characterization. a** Schematic illustration of Cu NSs and OD-Cu NSs synthesis. **b–e** SEM and HRTEM images of OD-Cu NSs (**b, d**) and Cu NSs (**c, e**), respectively. **f** Lattice spacings and of Cu NSs and

OD-Cu NSs. **g** XRD patterns. **h**, **i** The strain field maps of OD-Cu NSs (**h**) and Cu NSs (**i**) at E_{xy} , E_{xx} , and E_{yy} based on HRTEM images. Source data for Fig. 1 are provided as a Source Data file.

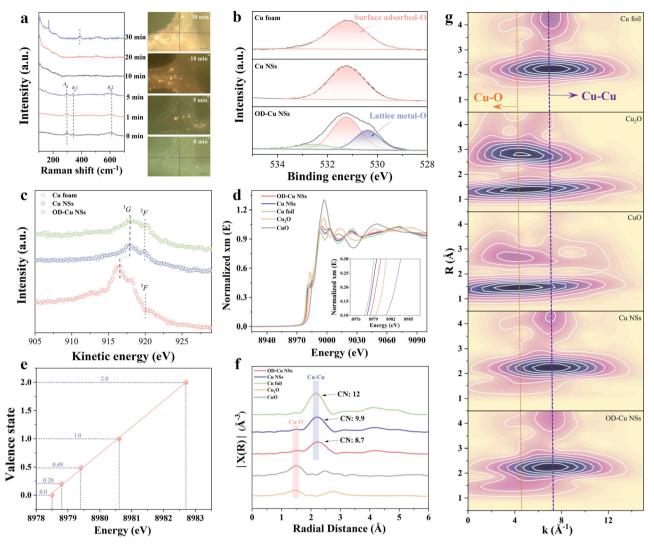


Fig. 2 | **Structural characterization. a** The operando Raman spectra (left panel) and time-resolved light microscopy images (right panel) of OD-Cu NSs surface at −0.87 V. **b** O 1 s XPS spectra and (**c**), Cu *L*₃*M*₄₅*M*₄₅ Auger spectra of Cu foam, Cu NSs, and OD-Cu NSs. **d** The Cu K-edge XANES spectra. **e** the fitted valence states. **f** the

corresponding k3-weighted FT-EXAFS spectra and (g), Morlet WT of the k3-weighted EXAFS data for Cu foil, Cu₂O, CuO, Cu NSs, and OD-Cu NSs. Source data for Fig. 2 are provided as a Source Data file.

during thermal reduction²⁹ (Fig. 1c and S3). High-resolution transmission electron microscopy (HRTEM) reveals both OD-Cu NSs and Cu NSs exhibit face-centered cubic (fcc) structures (Fig. 1d and e). Notably, the lattice spacing of the Cu (111) plane is 0.215 nm for OD-Cu NSs (Fig. 1f), which is larger than those of conventional Cu (0.208 nm) and Cu NSs (0.212 nm). The X-ray diffraction (XRD) patterns show the diffraction peaks of Cu NSs and OD-Cu NSs shift to lower angles in comparison to Cu foam (Fig. 1g), reflecting lattice tensile strains in Cu NSs and OD-Cu NSs as determined by the Bragg equation. The OD-Cu NSs have a lower diffraction angle than Cu NSs, which reveals the stronger tensile strains in OD-Cu NSs. The tensile strain value was calculated by the variation of crystal plane spacing. Compared to Cu foam, Cu NSs and OD-Cu NSs have tensile strains of 1.9% and 3.4%, respectively.

The spatial distribution of tensile strains in the catalysts was analyzed by geometric phase analysis. OD-Cu NSs exhibit substantial lattice strains across shear strain (E_{xy}), in-plane strain (E_{xx}), and out-of-plane strain (E_{yy}). Their strain intensities are significantly higher than those in Cu NSs (Fig. 1h and i). This is consistent with the results of XRD and HRTEM. The tensile strains of Cu NSs may originate from the new bond formation, bond breaking, or bond elongation/deflection during the formation of nanosheets³⁰. The more pronounced lattice strains in

OD-Cu NSs are likely due to the escape of oxygen atoms that bound to tetrahedral interstitial sites, leading to further lattice expansion.

Time-resolved operando Raman spectroscopy was used to monitor the oxidation state of the catalyst surface during electroreduction in the catalyst preparation process (Fig. 2a and Supplementary Fig. 4). The color of OD-Cu NSs surface gradually changes from its initial turquoise (CuO) to bright yellow (Cu) after reduction at -0.87 V (vs RHE) in Na₂SO₄ electrolyte. The Raman spectra at 0 min clearly show three peaks at 297, 343, and 614 cm⁻¹, corresponding to the A_g , B_g 1, and B_g 2 modes of CuO³¹. As the reduction proceeds, the intensities of CuO peaks gradually decrease until the CuO phase completely disappears at 10 min. Notably, a new band at around 390 cm⁻1 is observed as the reduction continues, which corresponds to a multi-phonon process of lattice oxygen in Cu. It indicates the presence of residual lattice oxygen and the possible formation of a new Cu suboxide phase³². Moreover, the O 1s XPS spectra (Fig. 2b) show that only surface-adsorbed oxygen is present in Cu foam and Cu NSs, whereas OD-Cu NSs exhibit a lattice metal-oxygen bond around 530.6 eV, confirming the presence of lattice oxygen in OD-Cu NSs³³. Its lattice oxygen content is estimated to be 10.08 at%. The valence state of Cu can be determined from the positions of ¹G and ³F peaks in the Cu Auger spectrum ($L_3M_{45}M_{45}$ transition). The 1G peaks for Cu²⁺, Cu⁺, and

Cu⁰ are usually located at 917.1, 915.8, and 918.0 eV, respectively. In addition, the 3F peak is prominent in the spectrum for Cu⁰ but not in the spectra for Cu₂O and CuO³⁴. As shown in Fig. 2c, the 1G peaks for Cu foam and Cu NSs are located at 918.0 eV with a distinct 3F peak, consistent with the characteristics of Cu⁰. Notably, the 1G peak for OD-Cu NSs shifts to approximately 916.5 eV, lower than those of Cu⁰ and Cu²⁺ but close to Cu⁺. Furthermore, the 3F peak related to Cu⁰ still appears in OD-Cu NSs. These results suggest that the valence state of OD-Cu NSs is Cu⁸⁺ (0 < δ <1).

X-ray absorption near-edge structure (XANES) spectroscopy was used to investigate the electronic structure and coordination environment of catalysts. As displayed in Fig. 2d, Cu NSs and Cu foil exhibit insignificant differences in the absorption edge and transition intensity, indicating Cu NSs and Cu have similar coordination environments. However, OD-Cu NSs have higher near-edge absorption energy and lower $1s \rightarrow 4p$ transition intensity in comparison to Cu foil and Cu NSs, revealing an abnormal d-p hybridization in the Cu atoms of OD-Cu NSs³⁵. It can be attributed to the reduced electron density of 4p orbitals caused by the strong electron-withdrawing property of residual lattice oxygen atoms³⁶. In addition, the near-edge absorption energy of OD-Cu NSs is lower than that of Cu₂O with a fitted valence state of 0.49, suggesting a mixed local environment for its Cu atoms (Fig. 2e). In the k3-weighted Fourier-transformed EXAFS (FT-EXAFS) spectra (Fig. 2f), a distinct Cu-O peak (1.5 Å) appears in OD-Cu NSs, confirming the presence of residual lattice oxygen. The peak intensity of the Cu-Cu bond (2.23 Å)³⁷ in OD-Cu NSs is significantly lower than those of Cu foil and Cu NSs, indicating that the Cu-Cu coordination unsaturation of OD-Cu NSs is increased due to the presence of lattice oxygen atoms. Moreover, the Cu-Cu peaks in OD-Cu NSs and Cu NSs are slightly shifted to a larger radial distance, consistent with the existence of lattice tensile strains²³. The Morlet wavelet transform (WT) plots show that both Cu NSs and OD-Cu NSs exhibit the highest peaks at ~7Å⁻¹ (Fig. 2g), corresponding to the Cu-Cu shell. OD-Cu NSs show a distinct Cu-O shell at ~4.5 Å⁻¹, further supporting the presence of residual lattice oxygen in OD-Cu NSs. The EXAFS fitting (Supplementary Fig. 5, 6 and Supplementary Table 1) indicates that the Cu-Cu coordination number (CN) in OD-Cu NSs is 8.7, significantly lower than those in Cu NSs (9.9) and Cu foil (12), which can be explained by that the residual lattice oxygen in OD-Cu NSs reduces the saturation of Cu-Cu coordination. In addition, the average coordination number of Cu-O shell in OD-Cu NSs is 1.0 ± 0.2 , revealing the partial saturation of Cu atoms by O atoms in OD-Cu NSs.

The eNO3RR performance of OD-Cu NSs

The eNO3RR performance was evaluated in an H-type cell under ambient conditions. Considering the practical scenarios, a neutral Na₂SO₄ electrolyte was selected. The linear sweep voltammograms (LSV) of OD-Cu NSs (Supplementary Fig. 7) show a distinct current response for 10 mM NO₃⁻ at potential more negative than -0.1 V (vs RHE), indicating its good activity for eNO3RR. NO₃⁻ conversion to NH₃ was further evaluated at -0.3 V to -0.8 V. The NH₃ was quantitatively detected by colorimetric method and validated by ¹H-NMR (Supplementary Fig. 8). Notably, the NH₃ yields of OD-Cu NSs are higher than those of Cu NSs and Cu foam under all the applied potentials (Fig. 3a). At -0.7 V, OD-Cu NSs produce NH₃ at the maximum yield of 0.38 mmol·h⁻¹·cm⁻², which is 1.4 times higher than that of Cu NSs, highlighting the significant role of lattice oxygen and tensile strain in enhancing eNO3RR. The FE_{NH}, of OD-Cu NSs exhibit a volcano-type trend with the variation of potential (Fig. 3b). At -0.3 V, a high concentration of nitrite is accumulated as *NO₂ can't be efficiently converted, resulting in low FE_{NH_2} (Fig. 3c). As expected, its FE_{NH_2} is significantly increased, and nitrite is rapidly converted at more negative potentials. At -0.6 V, the FE_{NH₃} of OD-Cu NSs is 99.7%, significantly higher than those of Cu NSs (74.2%) and Cu foam (44.2%). The FE of NO₂⁻ is only 1.1% for OD-Cu NSs, much lower than 25.3% for Cu NSs and 28.7% for Cu foam (Fig. 3d and Supplementary Fig. 9). The NO₂⁻ concentration of OD-Cu NSs is also significantly lower than those of Cu NSs and Cu foam (Supplementary Fig. 10). It suggests that OD-Cu NSs can effectively promote nitrite conversion and alleviate *NO₂ poison at the active sites. The generation of byproducts such as NH₂OH, N₂H₄, and NO_x is negligible for eNO3RR on OD-Cu NSs based on the results of online differential electrochemical mass spectrometry (DEMS) and colorimetric methods (Supplementary Fig. 11-14). Considering that OD-Cu NSs and Cu NSs have similar morphology and crystalline structure, the better performance of OD-Cu NSs can be contributed from its residual lattice oxygen and tensile strains, which increase the active sites and result in higher electrochemical active surface area (Supplementary Fig. 15, 16). The FE_{NH} of OD-Cu NSs decreases at more negative potentials due to the occurrence of hydrogen evolution. In addition, OD-Cu NSs are also efficient for NH3 production at lower electrolyte concentrations (0.05-0.1M) with FE_{NH_3} of 91.2-96.7% (Supplementary Fig. 17). To assess the denitrification performance of OD-Cu NSs, a batch test of NO₃⁻ conversion was conducted at -0.6 V. As shown in Fig. 3e, almost all the nitrogen (10 mM NO₃⁻) is converted to NH₃ after 75 min, and the concentrations of NO₃-N (0.80 ppm) and NO₂-N (4.16 ppm) are well below the World Health Organization's (WHO's) drinking water standards³⁸. The isotope labeling eNO3RR experiment was performed by using K15NO3 as a nitrogen source (Fig. 3f). The quantitative results for ¹⁵NH₄⁺ were almost identical to those for ¹⁴NH₄⁺ from ¹⁴NO₃⁻ reduction (Supplementary Fig. 18), which proves that NH₃ is generated from NO₃⁻ reduction instead of other nitrogen sources.

Nitrate concentrations usually vary from ten to more than a hundred millimole per liter in industrial wastewaters^{39,40}. It is crucial for the catalyst to exhibit excellent nitrate-to-ammonia conversion performance across a wide range of NO₃⁻ concentrations. Herein NH₃ synthesis performance of OD-Cu NSs was further explored under 10-140 mM NO₃. The LSV curves show that the current density remarkably increases with the rise of NO₃⁻ concentration (Fig. 3g), indicating the good eNO3RR performance of OD-Cu NSs across a broad range of NO₃⁻ concentrations. Its NH₃ synthesis from 10-140 mM NO₃⁻ was examined at 50 ~ 1500 mA·cm⁻², considering that FE_{NH₃} normally declines under high current density⁴¹. As shown in Fig. 3h, OD-Cu NSs maintain high FE_{NH3} across a wide range of NO₃ concentrations, exceeding 88.7% even at a current density above 1000 mA·cm⁻². Furthermore, both the NH₃ yield and denitrification capability are improved with increasing NO₃⁻ concentration and current density. At 140 mM NO₃⁻ and 1500 mA·cm⁻², the FE_{NHa} is 88.9% with NH₃ yield of $6.20 \text{ mmol} \cdot \text{h}^{-1} \cdot \text{cm}^{-2}$ (105.44 mg·h⁻¹·cm⁻²) and NO₃ removal rate of 91.87 mg·h⁻¹·cm⁻² NO₃-N. Notably, OD-Cu NSs outperform the recently reported electrocatalysts with higher NH₃ yield while keeping an equal or even higher FE_{NH}, under similar conditions (10–100 mM NO₃⁻, Supplementary Table. 2). Meanwhile, OD-Cu NSs keep high FE for NH₃ production during 14 h of NO₃⁻ reduction (10 mM, Fig. 3i and Supplementary Fig. 19), demonstrating the good durability of OD-Cu NSs for NH₃ synthesis. After the stability test, OD-Cu NSs retain the initial nanosheet morphology and lattice strains (Supplementary Fig. 20a, b). Furthermore, XPS and Cu $L_3M_{45}M_{45}$ Auger spectra prove the well-maintained lattice oxygen and Cu valence state after 14 cycles of eNO3RR, which reveals its good stability (Supplementary Fig. 20c, d). Meanwhile, in situ Raman spectra confirm the well-maintained lattice oxygen for OD-Cu NSs (Supplementary Fig. 21). The OD-Cu NSs are stable for eNO3RR without structure reconstruction.

The catalytic mechanism of OD-Cu NSs

The reduction of NO₃⁻ to NH₃ involves a complex 8 electrons and 9 protons transfer process. For Cu-based catalysts, there are significant challenges in the conversion and hydrogenation of intermediates such as *NO₂, leading to limited eNO3RR performance⁴². To investigate the

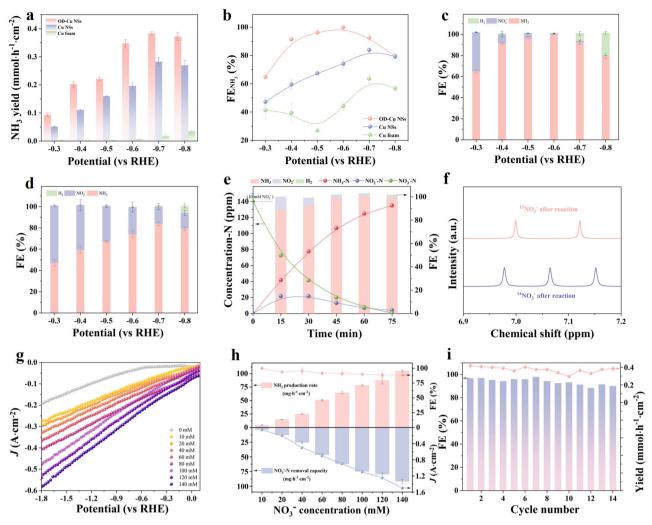


Fig. 3 | **Performance for eNO3RR. a** NH₃ yields and (**b**), corresponding FEs for 10 mM NO_3^- reduction on Cu foam, Cu NSs, and OD-Cu NSs. **c**, **d** The product distribution for OD-Cu NSs (**c**) and Cu NSs (**d**). **e** Time-dependent concentrations of NO₃⁻, NO₂⁻, and NH₃ during the reduction of 10 mM NO_3^- over OD-Cu NSs. **f** ¹H NMR spectra of samples from OD-Cu NSs catalyzed 15 NO₃⁻ or 14 NO₃⁻ reduction. **g** LSV

curves and (h), nitrate removal capacities and NH_3 production rates of OD-Cu NSs at 10–140 mM NO_3 . I Stability of OD-Cu NSs for 14 successive cycles of 10 mM NO_3 reduction at -0.6 V (Na_2SO_4 electrolyte). Error bars represent the standard deviation from three independent measurements. All the potentials have not been iR-corrected. Source data for Fig. 3 are provided as a Source Data file.

ability of OD-Cu NSs for the conversion of intermediates, the rate constants for NO₃⁻ conversion to NO₂⁻ (k₁, Supplementary Fig. 22, 23) and NO₂⁻ conversion to NH₃ (k₂, Supplementary Fig. 24, 25) were calculated to assess the reaction kinetics7. Meanwhile, the onset potentials for NO₃⁻ and NO₂⁻ reduction on OD-Cu NSs, Cu NSs, and Cu foam were compared (Supplementary Fig. 26). As can be seen from Fig. 4a, OD-Cu NSs have the highest k₁ and k₂ values among the three catalysts, which proves that OD-Cu NSs can realize rapid $NO_3^- \rightarrow NO_2^-$ and $NO_2^- \rightarrow NH_3$ conversion. Notably, OD-Cu NSs have the lowest onset potential for NO₂⁻ reduction, which further demonstrates its excellent NO₂ conversion performance. This may be attributed to the presence of residual lattice oxygen, which limits free electrons on the Cu surface and facilitates *NO₂ desorption and conversion (Fig. 4b, right panel). To illustrate the influence of lattice oxygen on *NO₂ conversion, OD-Cu with lattice oxygen and tensile strain of 1.9% (OD-Cu_{oxide}) was prepared (Supplementary Fig. 27a, b). The OD-Cu_{oxide} has the same tensile strain value as Cu NSs, but it exhibits significantly higher FE_{NH}, and lower nitrite accumulation than Cu NSs for eNO3RR under the same conditions (Supplementary Fig. 27c, d), demonstrating the crucial role of lattice oxygen in promoting *NO₂ conversion and NH₃ production. Although Cu NSs have a lower onset potential for NO₃⁻ reduction than OD-Cu NSs, its k₁ and k₂ values are lower than those of OD-Cu NSs. This may be caused by that the metallic property of Cu NSs favors NO_3^- reduction, but the generated NO_2^- poisons the active sites and reduces the reaction rate (Fig. 4b, left panel). For Cu foam, both NO_3^- and NO_2^- reduction require high overpotentials with the lowest k_1 and k_2 values. This is consistent with its worst eNO3RR performance.

The in situ ATR-FTIR spectroscopy was employed to probe the dynamic evolution of intermediates generated from eNO3RR on OD-Cu NSs (Supplementary Fig. 28). As shown in Fig. 4c, the characteristic peak at 1650 cm⁻¹ corresponds to the bending vibration of *H₂O in the thin electrolyte layer⁴³. Concurrently, the strong absorption peak at 3390 cm⁻¹ is attributed to the *O-H stretching. It implies that OD-Cu NSs are favorable for the adsorption and decomposition of water to produce *H. The band at 3250 cm⁻¹ can be assigned to *NH₂ bending⁴⁴. The intermediate at ~1430 cm⁻¹ is attributed to *NH₂OH, a key intermediate for NH₃ formation⁴³. Meanwhile, the downward band at 1240 cm⁻¹ belongs to *NO₂. As the potential is shifted from -0.3 V to -0.7 V, the peak intensity of *NO₂ increases gradually and suddenly flattens out at -0.7 V, indicating that *NO₂ undergoes a rapid conversion. It is noteworthy that the ratio of *NO₂ peak area to the total area of *NO₂ and *NH₂OH decreases sharply with the potential negatively shifted (Fig. 4d). Meanwhile, the peak intensity of *NH₃ (1100 cm⁻¹) exhibits a progressive enhancement at more negative potentials.

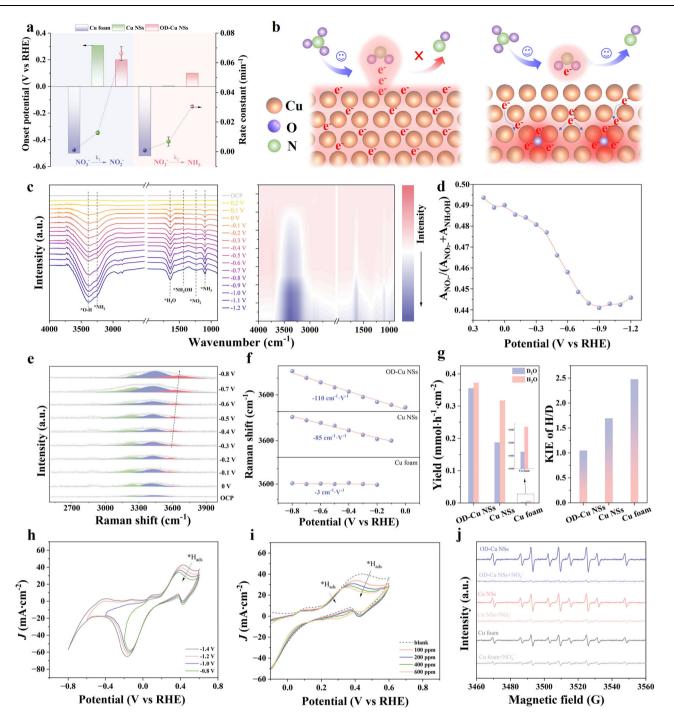


Fig. 4 | **Mechanism analysis for *NO₂ conversion and *H generation. a** The onset potentials and reaction constants for NO_3^- conversion to NO_2^- (k_1) and NO_2^- conversion to NH_3 (k_2) on OD-Cu NSs, Cu NSs, and Cu foam in $0.5\,M$ $Na_2SO_4+10\,mM$ NO_3^-/NO_2^- (Error bars represent standard deviation from three independent measurements). **b** Schematic diagram of lattice oxygen-mediated intermediate adsorption on Cu surface. **c** The operando ATR-FTIR spectra and 2D FTIR contour map. **d** the ANO₂/ANO₂ + ANH₂OH and (**e**), in situ Raman spectra of different interfacial water during eNO3RR on OD-Cu NSs at different potentials.

f Wavenumber shift of dangling O-H with potential changes. **g** NH $_3$ yields and KIE values for eNO3RR at -0.6 V using H $_2$ O and D $_2$ O as solvents. Cyclic voltammograms of OD-Cu NSs in (**h**), Ar-saturated 0.5 M Na $_2$ SO $_4$ under different starting potentials and (**i**), different NO $_3$ concentrations. **j** EPR spectra of *H signal for OD-Cu NSs, Cu NSs, and Cu foam with and without the presence of nitrate (-0.6 V). All the potentials have not been iR-corrected. Source data for Fig. 4 are provided as a Source Data file.

However, both Cu NSs and Cu foam exhibit distinct *NO₂ signals, and the peak intensity of *NO₂ almost keeps unchanged with the negative shift of potential (Supplementary Fig. 29). These results indicate that OD-Cu NSs are more efficient for converting *NO₂ to NH₃.

The *H supply from water dissociation is a key step for NH₃ production in neutral solution⁴⁵. In situ Raman spectroscopy was used to analyze the water peak at 2800 - 4000 cm⁻¹. By Gaussian fitting of the

interfacial water peak, tetrahedrally coordinated H_2O (tetra- H_2O , 3230 cm $^{-1}$), trihedrally coordinated H_2O (tri- H_2O , 3425 cm $^{-1}$) and dangling O-H bond (3600 cm $^{-1}$) of H_2O^{46} can be found. The dangling O-H is a weak hydrogen bond configuration, which tends to adsorb on the catalyst surface and dissociate to generate *H. Its position is red-shifted as the potential decreases (Fig. 4e), indicating preferential dissociation of dangling O-H on OD-Cu NSs 47 . A weaker shift of the dangling O-H

peak under more negative potential is observed on Cu NSs, and a negligible shift is observed on Cu foam compared to OD-Cu NSs (Supplementary Fig. 30, 31). It is noteworthy that the slopes of dangling O-H peak displacement show a trend of OD-Cu NSs (-110 cm⁻¹·V⁻¹) > Cu NSs (-85 cm⁻¹·V⁻¹) > Cu foam (-3 cm⁻¹·V⁻¹) (Fig. 4f), suggesting that OD-Cu NSs are more conducive to *H generation from water dissociation. OD-Cu NSs and Cu NSs display significantly enhanced ability for water dissociation to *H relative to Cu foam. Both Cu NSs and Cu foam don't contain lattice oxygen, but Cu NSs have tensile strains compared to Cu foam. It is reasonable to speculate that tensile strains play a key role in enhancing *H supply from water dissociation.

The kinetic isotope effect (KIE) of H/D (H₂O/D₂O) for eNO3RR was further examined to validate the role of lattice tensile strains in accelerating H2O dissociation. The KIE value is a good indicator of proton transfer kinetics, providing insights into the rate-determining step (RDS). KIE > 1 indicates that proton transfer is involved in the RDS, while KIE ≈ 1 suggests proton transfer is not a limiting factor⁴⁸. As shown in Fig. 4g, the NH₃ yield of Cu foam is significantly decreased after the replacement of H₂O by D₂O and its KIE value is 2.47, indicating the limited proton supply for eNO3RR. Cu NSs show a smaller change in NH₃ yield and current density after the replacement of H₂O by D₂O (Supplementary Fig. 32) and a KIE value of 1.69, suggesting that the proton supply capacity of Cu NSs is enhanced by the introduction of tensile strains. Interestingly, the NH3 yield and current density (Supplementary Fig. 33) of OD-Cu NSs almost keep unchanged after the replacement of H₂O by D₂O, and its KIE value is only 1.05. This phenomenon may be due to the fact that OD-Cu NSs have stronger tensile strains, which promotes water dissociation and provide sufficient *H for hydrogenation. Moreover, the linear relationship between tensile strain and KIE value further confirms that tensile strain accelerates *H generation (Supplementary Fig. 34). The *H generation ability of catalysts was investigated by measuring *Hads peak intensity via cyclic voltammogram at different initial potentials⁴⁹. OD-Cu NSs exhibit stronger *Hads peak intensity and larger intensity enhancement than Cu NSs as the applied potential decreases, whereas negligible *H_{ads} peak appears for Cu foam (Fig. 4h and Supplementary Fig. 35, 36). It further confirms the vital role of tensile strain in promoting *H generation. By increasing the NO₃⁻ concentration for cyclic voltammogram test, the *Hads intensity on OD-Cu NSs is reduced obviously (Fig. 4i), indicating that *H plays a crucial role in promoting NO₃⁻ reduction. Its good *H generation and utilization ability is also proved by EPR spectra (Fig. 4j). OD-Cu NSs exhibit stronger *H signal intensity than Cu NSs and Cu foam (-0.6 V). Its *H signal almost disappears after the addition of nitrate. In contrast, the *H consumption is slower for Cu NSs and Cu foam. It confirms the accelerated *H supply and utilization of OD-Cu NSs for nitrate reduction.

Density functional theory (DFT) calculation was employed to further elucidate the mechanism of residual lattice oxygen and lattice strain for regulating the eNO3RR activity of OD-Cu NSs. Based on the good matching between FT-EXAFS data of OD-Cu NSs and the simulated EXAFS result of atypical suboxide copper in previous works⁵⁰, we constructed the OD-Cu model (Cu₄O) (Supplementary data 1). To investigate the impact of residual lattice oxygen on eNO3RR, the tensile strain was not introduced into the computational model. The impact of residual lattice oxygen on the electronic state of the Cu surface was investigated (Supplementary Fig. 37). As shown in the electron localization function (ELF) map (Fig. 5a), the surface electrons on OD-Cu that contain residual lattice oxygen are more localized compared to those on Cu surface⁵¹. It implies that the valence electrons in OD-Cu are less free to move across the surface, thereby weakening the electronic coupling between the catalyst surface and *NO₂. This is conducive to reduce adsorption strength of *NO₂ intermediate. The applied potential has insignificant influence on the electronic structures of OD-Cu and Cu under the investigated conditions (Supplementary Fig. 38). The adsorption energy of *NO_2 on OD-Cu and Cu surface were further calculated (Supplementary Fig. 39). As depicted in Fig. 5b, the adsorption of *NO_2 on OD-Cu surface (-1.28 eV) is weaker than that on Cu surface (-1.62 eV), which is attributed to the strong electron-withdrawing effect of subsurface lattice oxygen on OD-Cu. It diminishes the electron transfer between its surface and the *NO_2 (Fig. 5c), thus alleviating the poison of active sites.

To explore the influence of tensile strain on *H generation from water dissociation, DFT calculation was conducted on the models of Cu (111) with 0%, 1.9%, and 3.4% lattice tensile strains (denoted as Cu-0%, Cu-1.9%, and Cu-3.4%) based on the tensile strains observed in OD-Cu NSs and Cu NSs (Supplementary Fig. 40–43 and Supplementary data 1). The water adsorption energy is -0.37 eV for Cu-0%, which decreases to -0.48 eV for Cu-1.9% and -0.82 eV for Cu-3.4%. Water adsorption is facilitated on the Cu surface with the rise of tensile strains. It may be due to the altered electronic structure by the increased tensile strains, leading to an upward shift of its d-band center (Supplementary Fig. 44). The upshifted d-band center strengthens the interaction between the catalyst surface and water, improving the adsorption of water molecules. The facilitated water adsorption favors subsequent water dissociation to produce *H. As shown in Fig. 5d, Cu-3.4% exhibits a low energy barrier ($\Delta G^*H_2O = 0.11 \text{ eV}$) for H_2O dissociation to *H, while ΔG^*H_2O increases to 0.22 eV for Cu-1.9% and 0.32 eV for Cu-0%, indicating that larger tensile strains are more favorable for *H generation. In addition, the protonation of the surface oxygen atoms of OD-Cu is thermodynamically unfavorable in comparison with eNO3RR (Supplementary Fig. 45), which is unlikely to influence *NO₂ conversion and water decomposition. Both theoretical and experimental results confirm that the electronic modulation mediated by residual lattice oxygen effectively promotes *NO2 conversion and mitigates active site poison. Concurrently, tensile strains improve the water dissociation capability of Cu, providing sufficient *H for hydrogenation. The synergistic effects of residual lattice oxygen and tensile strains enhance NH₃ synthesis on OD-Cu NSs (Fig. 5e).

To explore the reaction pathway for eNO3RR, the reaction intermediates were identified by in situ Raman and DEMS. In the Raman spectra (Supplementary Fig. 46), *NO₃ stretching peak is observed at 1050 cm⁻¹ under open circuit potential (OCP). When negative potential is applied, *NO₃ peak gradually disappears, while two new Raman bands appear at 1326 cm⁻¹ and 1591 cm⁻¹, corresponding to the antisymmetric stretching of *NO2 and *NH3, respectively. The signal intensity of *NO2 first increases and then disappears with the negative shift of potential. It reveals the generation of *NO2 and its transformation to ammonia during eNO3RR. The online DEMS shows a strong signal at m/z = 17, which is attributed to NH₃ (Supplementary Fig. 11). The other signals at m/z = 16, 33, 44, 15, and 30 can be assigned to NH₂,NH₂OH, N₂O, NH and NO, respectively. The signal intensities of these intermediates are 1-4 orders of magnitude lower than that of NH₃, which may be related to the rapid conversion of these intermediates. These results are consistent with in situ ATR-FTIR data in Fig. 4c. Based on the intermediates detected, the eNO3RR pathway and the reactionfree energies on OD-Cu and Cu surfaces were explored by DFT calculation. As shown in Fig. 5f, the reaction begins with the adsorption of NO₃ onto the catalyst surface, and the formed *NO₃ is spontaneously converted to *NO2 via N-O bond breaking. The *NO2 hydrogenation to *NO₂H is an uphill process with the free energy of 0.11 eV on the OD-Cu surface, significantly lower than that on the Cu surface (0.60 eV), highlighting the role of residual lattice oxygen in promoting *NO₂ conversion and suppressing nitrite accumulation. Subsequently, the *NO₂H is converted to *NO, followed by *NO reduction to *NOH. The formed *NOH is further reduced to *NH₃ via the pathway of *NOH → *NHOH \rightarrow *NH₂OH \rightarrow *NH₂ \rightarrow *NH₃ or *NOH \rightarrow *N \rightarrow *NH \rightarrow *NH₂ (downhill process). NH₃ is generated after *NH₃ desorption from the catalyst surface. Notably, the rate-determining step (RDS) on OD-Cu is *NO → *NOH. Its energy barrier is 0.37 eV, which is considerably lower than

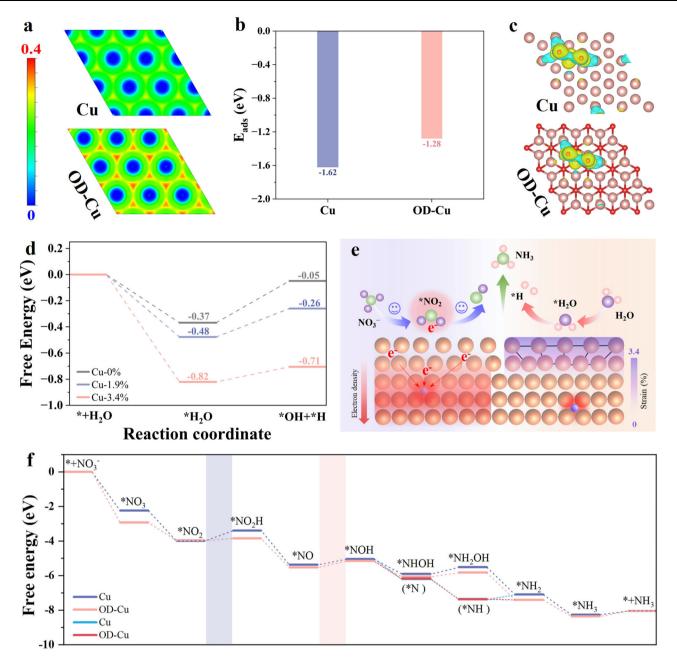


Fig. 5 | **Reaction mechanism from DFT calculation. a** ELF maps and **(b)** adsorption energies of NO₂⁻ on Cu and OD-Cu surfaces. **c** The adsorption model and charge density difference of Cu and OD-Cu (accumulation and deletion of electrons are shown in blue and yellow). **d** Gibbs free energy for H₂O dissociation to *H on Cu-

0%, Cu-1.9% and Cu-3.4% surfaces. **e** Schematic diagram of eNO3RR promoted by lattice oxygen-mediated electron regulation and lattice strain-mediated water dissociation. **f** Gibbs free energy diagrams for eNO3RR on the surfaces of Cu and OD-Cu. Source data for Fig. 5 are provided as a Source Data file.

0.60 eV for the RDS of Cu (*NO $_2$ > *NO $_2$ H). Meanwhile, the effect of tensile strains on reducing the energy barrier of RDS is insignificant for OD-Cu (Supplementary Fig. 47). These results demonstrate that the lattice oxygen of OD-Cu significantly reduces the energy barrier for eNO3RR and promotes NH $_3$ production.

Application for ammonia recovery

Given the excellent eNO3RR performance of OD-Cu NSs, the conversion of NO $_3$ ⁻ into high-purity, ammonia-based chemicals was explored to assess its potential for practical application. NH $_3$ in the electrolyte after eNO3RR (100 mM NO $_3$ ⁻, $-1.1\,\mathrm{A\,cm^{-2}}$) was extracted via argon stripping at 50 °C (Fig. 6a). Approximately 95.0% of NH $_3$ vapor is successfully removed from the electrolyte (Fig. 6b), indicating the system is effective for NO $_3$ ⁻ removal and NH $_3$ recovery. 92.2% of the stripped

NH₃ vapor is captured through an acidic solution (HCl), and 82.9% of high-purity NH₄Cl powder is collected after solvent evaporation (Fig. 6c and Supplementary Fig. 48). This demonstrates the feasibility of OD-Cu NSs-based eNO3RR system for NH₃ recovery. For eNO3RR under industrial-scale current density, slow mass transfer and incontiguous NH₃ production in static H-cell hinder its large-scale application. Flow cells can enhance mass transfer and provide a stable reaction microenvironment (e.g., local concentration and pH) for continuous NH₃ production compared to H-cell. Therefore, we assembled the OD-Cu NSs catalyst into a membrane electrode assembly (MEA) to investigate its performance and long-term stability for NO₃⁻ reduction to NH₃ under industrial-scale current. The setup of MEA is illustrated in Supplementary Fig. 49, 50. The total current for long-term operation was set at 1 A. As shown in Fig. 6d, OD-Cu NSs still

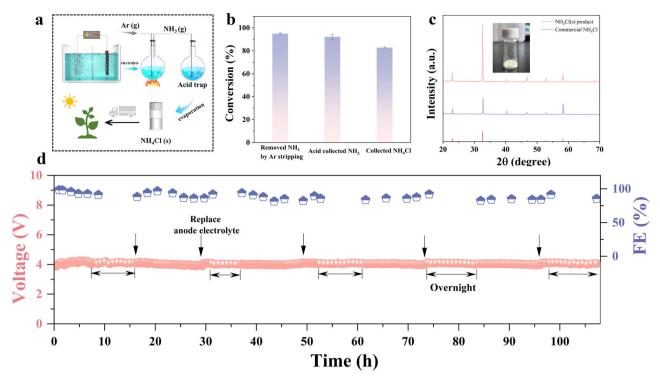


Fig. 6 | **Ammonia recovery and stability. a** Schematic illustration of the process for converting nitrate to high-purity NH₄Cl. **b** The conversion efficiency for NH₃ synthesis from the reduction of 100 mM NO₃ $^-$ at -1.1 A cm $^{-2}$ (Error bars represent standard deviation from three independent measurements). **c** XRD pattern of the

synthesized NH₄Cl product. **d** Long-term stability for eNO3RR over OD-Cu NSs using a MEA system (30 mM NO $_3$ $^-$, 1.0 A, 4.0 cm 2 , Na $_2$ SO $_4$ electrolyte). Source data for Fig. 6 are provided as a Source Data file.

achieve a FE $_{\rm NH_3}$ of 98.6% and NH $_3$ yield of 1.15 mmol·h $^{-1}$ ·cm $^{-2}$ in the MEA system (30 mM NO $_3$). The FE $_{\rm NH_3}$ and NH $_3$ yield remain $_7$ 90% of initial values with negligible potential decay after 100 h of operation, demonstrating the good stability of OD-Cu NSs for NH $_3$ synthesis. Techno-economic analysis shows that NH $_3$ electrosynthesis from nitrate wastewater is economically and technically feasible on OD-Cu NSs-based MEA system (Supplementary Fig. 51 and Supplementary Note 1). The effluent from the MEA was subjected to NH $_3$ separation and conversion. The mass flow and product distribution of eNO3RR are shown in Supplementary Fig. 52. After eNO3RR, the nitrate removal rate reaches 98%, and 84% of nitrate is converted to NH $_4$ Cl, illustrating its applicability for scaled applications in nitrate wastewater treatment and resource utilization.

Discussion

OD-Cu NSs with residual lattice oxygen and lattice tensile strains were prepared by a simple method. The OD-Cu NSs demonstrated remarkable eNO3RR activity and stability with nearly 100% FE_{NH_3} at 10 mM NO_3^- and neutral solution. It was also efficient for NH $_3$ electrosynthesis with FE_{NH_3} of 88.7–99.7% and maximum NH $_3$ yield of 6.20 mmol·h $^{-1}$ ·cm $^{-2}$ across a wide range of NO_3^- concentrations (10–140 mM) at 50–1500 mA·cm $^{-2}$. It converted over 80% of 100 mM NO_3^- into highpurity NH $_4$ Cl. The superior performance of OD-Cu NSs was attributed to electron regulation mediated by residual lattice oxygen and water dissociation mediated by lattice strains, which accelerated *NO $_2$ conversion and *H supply for NH $_3$ synthesis from eNO3RR. This work broadens the mechanistic understanding and will open up new possibilities for enhancing electrocatalytic performance in other fields.

Methods

Chemicals

Cu foam (thickness: 2.0 mm) was purchased from Suzhou Keshenghe Metal Material Co., Ltd. Ammonium persulphate ((NH₄)₂S₂O₈, AR 98%)

and Sodium hydroxide (NaOH, AR 99.9%) were obtained from Aladdin. All the chemicals were received and used without further purification. Deionized water (18.25 m Ω -cm $^{-1}$) was used in all experiments.

Synthesis of OD-Cu NSs, Cu NSs and OD-Cu_{oxide}

In a typical procedure, Cu foam was ultrasonically treated with 1M hydrochloric acid and anhydrous ethanol for 15 min each to remove surface impurities. The Cu foam was then rinsed thoroughly with ultrapure water, dried under vacuum, and stored for later use. To prepare the oxidation solution, 16 g of sodium hydroxide was dissolved in ultrapure water (70 mL). Subsequently, 3.2 g of ammonium persulfate was added to the sodium hydroxide solution and dissolved using ultrasound treatment. The Cu foam was immersed in the prepared solution and oxidized at 60 °C for 20 min. After oxidation, the Cu foam was rinsed with ultrapure water, and CuO nanosheets were obtained. The OD-Cu NSs were prepared by electrochemical reduction of CuO nanosheets for 30 min in 0.5 M Na₂SO₄ electrolyte at -0.87 V vs RHE (All the potentials have not been iR-corrected in this work). As comparison, Cu NSs were prepared by annealing CuO nanosheets at 350 °C for 2 h under 10% H₂/Ar atmosphere. OD-Cu_{oxide} was prepared through direct pyrolysis of copper foam (450 °C for 2 h under Ar), followed by electrochemical reduction under the same conditions as OD-Cu NSs preparation.

Characterization

The morphology and structure of catalysts were examined using a scanning electron microscope (Hitachi S4800) and a transmission electron microscope (FEI Talos F200S). X-ray photoelectron spectroscopy (XPS) data were acquired with an ESCALAB 250Xi spectrometer. All spectra were calibrated by setting the C1s peak to 284.8 eV. Raman spectra were recorded on a Renishaw inVia Qontor confocal Raman spectrometer using a 633 nm laser. X-ray diffraction (XRD) patterns were collected on a Rigaku Ultima IV diffractometer. ¹H NMR spectra

were obtained using a BRUKER AVANCE NEO 600 M spectrometer. Ultraviolet-visible (UV-Vis) absorbance spectra were measured with a SP-756P spectrophotometer. The TPR- H_2 experiment was carried out on an Autochem II 2920-MKS CirrusTM 3.

In situ, Raman measurements were performed using a customized Raman reaction cell and a CHI 760E electrochemical workstation. The conditions for the Raman test were the same as those of electrochemical tests, except that the electrolytes (50 mL) for the cathode and anode chambers were circulated outside the cell using a peristaltic pump.

In situ attenuated total reflection Fourier-transform infrared (ATR-FTIR) spectroscopy was conducted using a Thermo Scientific iS50 FT-IR instrument equipped with a mercury cadmium telluride (MCT) detector. The experiments employed a custom-made infrared cell connected to a CHI 760E electrochemical workstation. The electrolyte solution consisted of 0.5 M $\rm Na_2SO_4$ and 0.1 M KNO₃. After subtracting the infrared background, spectra were recorded under applied potentials with Ag/AgCl as the reference electrode and platinum (Pt) as the counter electrode. The applied potential was from 0.2 V to -1.2 V.

The H_2 -TPR experiment was conducted on an Autochem II instrument. OD-Cu NSs were loaded into a quartz tube and pretreated by sweeping with Ar at 120 °C for 30 min to remove any adsorbed species. After cooling to 25 °C, the temperature was increased to 600 °C at a rate of 10 °C min⁻¹ under a gas mixture of 10% H_2 /Ar. The H_2 consumption during the process was continuously monitored and quantified using a thermal conductivity detector (TCD).

Electrochemical experiments

The eNO3RR was conducted in a customized H-type cell using a CHI1140C electrochemical workstation. The anodic and cathodic compartments were separated by a Nafion 117 membrane (DuPont). The Nafion 117 membrane was soaked in 5 wt% H₂O₂ solution at 80 °C for 1 h, followed by immersing in deionized water for 30 min. It was then treated with 0.5 M H₂SO₄ at 80 °C for 1 h and rinsed with deionized water for another 30 min before use. The as-synthesized electrocatalyst served as the working electrode. An Ag/AgCl electrode was used as the reference electrode, and a platinum plate was used as the counter electrode. $0.5 \, \text{M} \, \text{Na}_2 \text{SO}_4$ solution (pH = 6.8 ± 0.3) was used as the electrolyte for both the cathode and anode chambers, and 10-140 mM NO₃ was added to the cathode chamber. The solution in the cathode chamber was stirred during the process of eNO3RR. All potentials were measured against the Ag/AgCl reference electrode and converted to the reversible hydrogen electrode (RHE) scale using the equation: $E(vs. RHE) = E(vs. Ag/AgCl) + 0.1976 V + 0.0591 \times pH.$

Analytical methods

The concentrations of $\mathrm{NH_4}^+$, $\mathrm{NO_3}^-$, and $\mathrm{NO_2}^-$ in the electrolyte (diluted to the appropriate concentration) before and after eNO3RR were determined via a colorimetric method.

NH₄ $^+$ detection: NH₄ $^+$ was determined by the Nessler's reagent method. $100\,\mu$ L NaKC₄H₄O₆ solution ($500\,g$ /L) and $100\,\mu$ L Nessler's reagent were added into a 5 mL sample and mixed thoroughly. The absorbance was determined at 420 nm after 20 min.

 NO_2^- detection: $100\,\mu l$ of color developer (0.04 g of $C_{12}H_{14}N_2\cdot 2HCl$, 0.8 g of sulfonamide, and 2 mL of 85% H_3PO_4 dissolved in 10 mL water) was added to 5 mL of diluted sample, and the reaction was carried out for 10 min at room temperature (20–25 °C). The absorbance was measured at 540 nm.

 NO_3^- detection: 1.1 mL of color developer (1M HCl and 0.8 wt% $NH_2SO_3H)$ was added to 5 mL of diluted sample. The absorbance was measured at 220 nm and 275 nm after 5 min. The final absorbance value was obtained by the equation of A = $A_{220nm}\text{-}2A_{275nm}$.

 N_2H_4 detection: The color reagent is consisted of 5.99 g of 4-dimethylaminobenzaldehyde ($C_9H_{11}NO$), 30 mL of concentrated

HCl, and 300 mL of ethanol. The color reagent was added into sample solution (v:v = 1:1), and the mixture was stirred for 10 min at room temperature (20–25 °C). The absorbance was measured at 457 nm. The calibration curve was obtained from N_2H_4 solutions (in 0.1 mol L^{-1} HCl) with known concentrations.

NH $_2$ OH detection: 100 μ L of acetate buffer (1.0 M sodium acetate +1.0 M acetic acid), 100 μ L, 4 mM ammonium ferric sulfate solution, and 100 μ L, 10 mM 1, 10-phenanthroline ethanolic solution were sequentially added into 3 mL of sample, and the mixture was stood for 20 min at room temperature (20–25 °C). The absorbance was measured at 510 nm.

Isotope labeling experiments

The $K^{15}NO_3$ was used as an isotope-labeled N source for eNO3RR. For 1H NMR (BRUKER AVANCE NEO 600 M) quantification, the pH value of the sample (25 mL) after eNO3RR was adjusted to 2 by adding H_2SO_4 . The $C_4H_4O_4$ (0.01 g) was added as an internal standard. 50 µL of DMSO-d6 was added into 0.5 mL of the above solution before 1H NMR test. The calibration curve was obtained using the peak area ratio of ${}^{15}NH_4{}^{+}.{}^{15}N$ to $C_4H_4O_4$ with 50, 100, 150, 200 mg L^{-1} of ${}^{15}NH_4{}^{+}$ solutions.

Ammonia yield and Faradaic efficiency

 FE_{NH_3} was obtained by Eq. (1), and the NH₃ yield was determined via Eq. (2), FE of NO_2^- was calculated by Eq. (3).

$$FE_{NH3}(\%) = (8F \times C_{NH3} \times V)/(M \times Q) \tag{1}$$

$$Yield_{NH3}(mmol \cdot h^{-1}cm^{-2}) = (C_{NH3} \times V)/(M \times t \times I)$$
 (2)

$$FE_{NO_{2}^{-}}(\%) = (2F \times C_{NO_{2}^{-}} \times V)/(46 \times Q)$$
 (3)

Where F is the Faraday constant (96485 C mol⁻¹), CNH₃ is NH₃ concentration (mol L⁻¹), V is the volume of catholyte (L), Q is the quantity of charge (C), M is the relative molecular mass of NH₃, t is the reaction time (s), I is the current (A), CNO₂⁻ is the NO₂⁻ concentration (mol L⁻¹).

Density functional theory (DFT) calculation

All the calculations were performed with the projector-augmented plane-wave method implemented in the Vienna ab initio simulation package. The generalized gradient approximation proposed by Perdew-Burke-Ernzerhof (PBE) was selected for the exchange-correlation potential. The cut-off energy for the plane wave was set to 480 eV. The energy criterion was set to 10^{-4} eV in the iterative solution of the Kohn-Sham equation. All the structures were relaxed until the residual forces on the atoms have declined to less than 0.02 eV/Å. To avoid interlaminar interactions, a vacuum spacing of 20 Å is applied perpendicular to the slab. VASPsol code with an implicit solvation model was used to simulate the electrolyte solution to deal with the solvation effect. Here, differences in Gibbs free energy (ΔG) (Eq. 4) for intermediates in eNO3RR defined as:

$$\Delta G = \Delta E + \Delta E_{ZPE} - T\Delta S + \Delta G_U \tag{4}$$

where ΔG is the total energy difference between the slab and respective terminations computed by DFT-PBE. ΔE_{ZPE} and T ΔS denote differences in zero-point energy and entropy between adsorbed states of reaction intermediates and gap phase, respectively. T is the room temperature (20–25 °C). ΔG_U = –eU, in which U is the electrode potential.

Data availability

The data supporting the conclusions of this study are presented in the paper and the supplementary information. Source data are provided in this paper.

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Author contributions

Q.W. and Y.L. conceived the idea. Q.W., X.F., and Y.L. designed the experiments. Q.W. performed the experiments. L.Q. helped with material characterization. Q.W., X.F., B.S., and X.Q. analyzed the data and discussed the results. Q.W. and Y.L. co-wrote the manuscript. All authors commented on the manuscript. Y.L. supervised the whole project.

Competing interests

The authors declare no competing interests.

Additional information

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