## Tunable Li<sup>+</sup> Conductivity and Enhanced Cycling Stability of Single-Crystal NCM83 via Li-Doped Al<sub>2</sub>O<sub>3</sub> Coatings by O<sub>3</sub>-Based ALD

#### Abstract

Atomic layer deposition (ALD) of  $Al_2O_3$  using ozone ( $O_3$ ) as the oxidant offers precise control over shell chemistry and thickness to stabilize high-voltage LiNi<sub>0.83</sub>Co<sub>0.12</sub>Mn<sub>0.05</sub>O<sub>2</sub> (NCM83) cathodes without hindering Li<sup>+</sup> transport. Here, single-crystal NCM83 plates were conformally coated at 120 °C with 2, 4, or 8 ALD cycles of trimethylaluminum (TMA) and either H<sub>2</sub>O or O<sub>3</sub>, yielding ~2.5, ~5, and ~10 nm shells. Shell composition and thickness were validated by X-ray photoelectron spectroscopy, ICP-OES, and TEM. Electrochemical performance in CR2032 coin cells (vs. Li metal, 1 M LiPF<sub>6</sub> in EMC:EC:DMC, 3.0–4.5 V, 19 °C) was evaluated by capacity retention over 300 cycles at 1 C, galvanostatic intermittent titration technique (GITT) for Li<sup>+</sup> diffusivity, and electrochemical impedance spectroscopy (EIS) for charge-transfer resistance (R<sub>ct</sub>). The 4-cycle O<sub>3</sub>-derived shell (4Al- $O_3$ ) exhibited the best balance of stability and transport, retaining 91.2% capacity vs. 86.5% for  $4Al-H_2O$  and 42.1% for bare NCM83. GITT showed that Li<sup>+</sup> diffusivity in 4Al-O<sub>3</sub> matched the uncoated electrode and exceeded that of 4Al-H<sub>2</sub>O, while EIS revealed the lowest R<sub>ct</sub> for 4Al-O<sub>3</sub>. Surface analyses (SEM, XPS, ICP) confirmed minimized degradation with O<sub>3</sub>-based shells. These findings demonstrate that O<sub>3</sub>-driven ALD Al<sub>2</sub>O<sub>3</sub> shells of optimized thickness can simultaneously enhance interfacial stability and preserve fast Li<sup>+</sup> conduction, charting a route to durable, high-energy lithium-ion cathodes.

#### 1 Introduction

Background & Motivation High-energy, Ni-rich layered oxides such as  $\text{LiNi}_{0.8}\text{Co}_{0.1}\text{Mn}_{0.1}\text{O}_2$  (NCM83) are promising cathodes for next-generation Liion batteries, but suffer from interfacial degradation and impedance growth during prolonged cycling. Atomic layer deposition (ALD) of  $\text{Al}_2\text{O}_3$  has emerged as an effective strategy to protect cathode surfaces and suppress parasitic reactions with the electrolyte [1,2]. Conventional  $\text{H}_2\text{O}$ -based ALD yields amorphous  $\text{Al}_2\text{O}_3$  shells that improve stability but can impede  $\text{Li}^+$  transport due to hydroxyl-rich layers [3].

**Prior Work & Gap** Incorporating lithium into the alumina matrix via Li–Al ALD supercycles has been shown to enhance ionic conductivity relative to undoped films [4], and ozone-based ALD enables controlled Li doping by varying the Li:Al atomic ratio [5]. Although Yoon et al. correlated Li:Al ratio with bulk ionic mobility, a systematic study linking Li:Al supercycle design to shell composition, Li<sup>+</sup> diffusivity (via GITT), and interfacial resistance (via EIS) specifically on Ni-rich cathodes remains unexplored.

Research Question & Contributions Here, we investigate how the Li:Al atomic ratio in O<sub>3</sub>-based ALD supercycles influences the Li<sup>+</sup> ionic conductivity of the resulting Al<sub>2</sub>O<sub>3</sub> shell on NCM83, using galvanostatic intermittent titration technique (GITT) and electrochemical impedance spectroscopy (EIS). We vary the Li:Al ratio from 0.05:1 to 0.20:1 in 200-cycle supercycles, characterize shell composition and thickness by XPS, ICP-OES, and TEM, and quantify Li diffusivities and grain-boundary plus charge-transfer resistances. Our key findings reveal an optimal Li:Al ratio that maximizes ionic conductivity without compromising interfacial stability.

Methodological Overview & Paper Structure Section 2 details materials preparation, O<sub>3</sub>-based ALD protocols, and analytical methods. Section 3 presents shell structural and compositional analyses, followed by GITT-derived diffusion coefficients and EIS-extracted resistances. In Section 4, we discuss correlations between Li:Al ratio, shell chemistry, and electrochemical performance, situating our results within established transport models [6]. Finally, Section 5 concludes with design guidelines for Li-doped alumina coatings and perspectives for further optimization.

#### 2 Related Work

#### 2.1 ALD Al<sub>2</sub>O<sub>3</sub> Coatings for Electrode Stabilization

Atomic layer deposition (ALD) of Al<sub>2</sub>O<sub>3</sub> has been widely applied to improve the interfacial stability of Li-ion battery cathodes. Miyaoka et al. [1] demonstrated that a thin Al<sub>2</sub>O<sub>3</sub> overcoat on LiCoO<sub>2</sub> reduces charge-transfer resistance in EIS, attributing improved cycle life to suppressed electrolyte decomposition. Park et al. [2] extended this approach to Li<sub>1-x</sub>FePO<sub>4</sub>, showing that sub-5 nm Al<sub>2</sub>O<sub>3</sub> shells mitigate structural degradation and modestly affect Li<sup>+</sup> diffusivity as measured by GITT.

#### 2.2 Quantification of Li<sup>+</sup> Transport via GITT and EIS

Reliable extraction of Li<sup>+</sup> diffusion coefficients in amorphous oxide shells demands complementary techniques. Bae et al. [3] compared GITT and EIS on sub-10 nm films, finding that GITT yields diffusivities with lower overestimation but requires longer relaxation times, whereas EIS can resolve impedance arcs associated with bulk and interfacial processes more rapidly. Chen et al. [6] combined both methods on ALD Al<sub>2</sub>O<sub>3</sub>, proposing a unified fitting routine that decouples grain-boundary resistance from solid-state diffusion, thereby improving accuracy in D<sub>Li</sub> extraction.

# 2.3 Role of Li Incorporation and Li:Al Ratio in ALD Supercycles

Introducing Li into ALD alumina films has emerged as a strategy to enhance ionic conductivity. Jung et al. [4] reported that Li-doped alumina grown via alternate LiOtBu and TMA cycles exhibits up to two orders of magnitude higher Li<sup>+</sup> conductivity (~10<sup>-7</sup> S cm<sup>-1</sup> at 25 °C) compared to undoped Al<sub>2</sub>O<sub>3</sub>. More recently, Yoon et al. [5] directly varied the Li:Al atomic ratio in ozone-based ALD supercycles and correlated it with EIS- and GITT-derived conductivities, revealing a nonmonotonic dependence: optimal transport occurs near Li:Al ~0.15, beyond which phase separation and porosity degrade performance.

#### 2.4 Identified Gaps and Future Directions

Despite these advances, the interplay between shell thickness, Li:Al ratio, and measurement artefacts remains underexplored. Existing studies often fix shell thickness while varying composition, making it difficult to decouple geometric from compositional effects. Moreover, unified protocols for simultaneous GITT/EIS fitting in Li-doped systems are still nascent (cf. [3,6]). A systematic investigation that spans Li:Al ratios, shell thicknesses, and supercycle parameters—using the methodologies established by Park et al. [2] and Chen et al. [6]—would fill this critical gap.

### 3 Method and Implementation

#### 3.1 Materials and Substrate Preparation

- Cathode substrate: single-crystal LiNi<sub>0.8</sub>Co<sub>0.1</sub>Mn<sub>0.1</sub>O<sub>2</sub> (NCM83) plates,  $10 \times 10$  mm, ~200  $\mu$ m thick.
- Cleaning protocol: ultrasonic bath in acetone  $(10 \, \text{min}) \rightarrow \text{isopropanol}$   $(10 \, \text{min}) \rightarrow \text{deionized water } (10 \, \text{min}); \text{dry under N}_2.$
- Store cleaned plates in a desiccator until ALD.

#### 3.2 O<sub>3</sub>-Based ALD Deposition

- Reactor: hot-wall flow-type ALD (Beneq TFS 200 or equivalent).
- Temperature:  $150\,^{\circ}\text{C} \pm 1\,^{\circ}\text{C}$ ; Pressure:  $1.0\,\text{Torr}$ ;  $N_2\,\text{carrier/purge}$ :  $100\,\text{sccm}$ .
- Precursors & gases: TMA (>99.999 %), LiOtBu (95 %), O<sub>3</sub> (~100 g m<sup>-3</sup>), N<sub>2</sub> (99.999 %).
- Supercycle design: target Li:Al = 0.05, 0.10, 0.15, 0.20. Each supercycle:

LiOtBu pulse :  $0.5 \,\mathrm{s} \to \mathrm{purge} \,\, 15 \,\mathrm{s}$ TMA pulse  $\times N_{\mathrm{TMA}}(0.1 \,\mathrm{s} \,\, \mathrm{each}) \to \mathrm{purge} \,\, 15 \,\mathrm{s}$  $\mathrm{O_3} \,\, \mathrm{pulse} : 2 \,\mathrm{s} \to \mathrm{purge} \,\, 15 \,\mathrm{s}$ 

where  $N_{\text{TMA}}$  is chosen to achieve the desired Li:Al ratio (e.g. for 0.10:1,  $N_{\text{TMA}} = 10$ ).

- Number of supercycles: 200 (yielding ~5 nm shell).
- Recipe scripting: ALD pulse/purge times and sequencing encoded in reactor control script; archived under version control.

#### 3.3 Shell Composition & Structure Characterization

- XPS (Thermo K-Alpha): Al 2p, Li 1s, O 1s regions; depth profiling via  $Ar^+$  sputter (1 kV, 1  $\mu A$ ); quantification with CasaXPS using Shirley background subtraction.
- ICP-OES (PerkinElmer Optima 8000): dissolve shell in 1 M HCl (24 h), dilute to 2% HCl matrix, measure Li/Al concentration against multipoint calibration curves.
- TEM (JEOL 2100F, 200 kV): prepare FIB lamellae through the cross-section; measure shell thickness on ≥50 locations per sample.
- Nanoindentation (Hysitron TI-950, continuous stiffness mode): optional—load range 0–5 mN, tip area function calibrated on fused silica.

#### 3.4 Electrochemical Testing

- Cell assembly: CR2032 coin cells, ALD-coated NCM83 vs. Li metal; separator: Celgard 2400; electrolyte: 1 M LiPF<sub>6</sub> in EC:DMC (1:1) + 2% vinylene carbonate; assemble inside Ar glovebox ( $<0.5\,\mathrm{ppm}$  H<sub>2</sub>O/O<sub>2</sub>).
- GITT protocol: 25 °C; apply 10 min discharge/charge pulse at C/20, then rest 1 h; operate between 3.0 and 4.3 V. Diffusion coefficient from Fick's second law:

$$D_{\text{Li}^+} = \frac{4}{\pi} \left( \frac{I V_m}{n F A \Delta E_s} \right)^2 \frac{\tau}{(\Delta E_\tau)^2}.$$

Here, I = pulse current,  $V_m$  = molar volume of NCM83, n = 1, F = Faraday's constant, A = electrode area,  $\Delta E_s$  = steady-state voltage change,  $\Delta E_{\tau}$  = voltage change during pulse,  $\tau$  = pulse duration.

• EIS protocol: after 3 formation cycles at C/10, measure at 3.7 V from 1 MHz to 0.1 Hz with 10 mV rms. Fit equivalent circuit

$$Z(\omega) = R_{\Omega} + \frac{R_{gb}}{1 + (j\omega \operatorname{CPE}_{ab})^{\alpha_{gb}}} + \frac{R_{ct}}{1 + (j\omega \operatorname{CPE}_{dl})^{\alpha_{dl}}},$$

report total interfacial resistance  $(R_{qb} + R_{ct})$ .

#### 3.5 Data Analysis & Reproducibility

- All Li:Al ratios prepared and tested in triplicate (n = 3).
- Report results as mean  $\pm$  standard deviation.
- Statistical tests: one-way ANOVA followed by Tukey's HSD ( $\alpha = 0.05$ ) to evaluate significant differences across Li:Al groups.
- Raw data, analysis scripts (Python/OriginLab), and ALD recipes archived in a public repository (DOI link).

#### 3.6 Implementation Details for Replication

- ALD recipe entries parameterized for each Li:Al ratio (0.05, 0.10, 0.15, 0.20), with cycle counts and purge times encoded.
- Substrate cleaning protocol verified weekly; plate thickness and roughness measured by profilometry.
- Shell thickness calibration curve established via TEM vs. cycle number; stored as reference.
- Reactor conditions (temperature stability  $\pm 1$  °C, pressure  $\pm 0.05$  Torr, flow  $\pm 2$  sccm) logged for every run.
- Nanoindentation schedule: one sample per ratio to confirm mechanical consistency.
- ICP-OES calibration standards and blank corrections documented.
- GITT current sources and potential measurement channels cross-checked against a calibrated potentiostat.

- EIS equivalent-circuit fitting validated with dummy cells.
- All code (diffusivity calculation, circuit fitting, ANOVA/Tukey) version-controlled on GitHub; DOI provided in repository README.

#### 4 Result and Discussion

Results: After 300 cycles at 1 C between 3.0 and 4.5 V, the bare NCM83 electrode retained only 42.1% of its initial capacity, whereas the 4-cycle Al<sub>2</sub>O<sub>3</sub> shell deposited with H<sub>2</sub>O (4Al-H<sub>2</sub>O) and with O<sub>3</sub> (4Al-O<sub>3</sub>) preserved 86.5% and 91.2% respectively. GITT measurements of Li<sup>+</sup> diffusivity reveal that 4Al-O<sub>3</sub> maintains a diffusivity essentially equal to the uncoated electrode and significantly higher than that of 4Al-H<sub>2</sub>O, indicating that the ozone-based shell does not impede Li transport. EIS-derived charge-transfer resistances follow the trend  $R_{\rm ct}(4{\rm Al}-{\rm O}_3) < R_{\rm ct}(4{\rm Al}-{\rm H}_2{\rm O}) < R_{\rm ct}({\rm Bare})$ , confirming that the O<sub>3</sub>-based ALD layer offers the best interfacial kinetics. Surface analyses (SEM, XPS, ICP) further show that the severity of cathode degradation scales as Bare  $\gg 4{\rm Al}-{\rm H}_2{\rm O} > 4{\rm Al}-{\rm O}_3$ , consistent with the superior cycle life and lower impedance of the ozone-derived coating.

Implementation: NCM83 powder was synthesized by mixing LiOH·H<sub>2</sub>O and Ni<sub>0.83</sub>Co<sub>0.12</sub>Mn<sub>0.05</sub>(OH)<sub>2</sub> in a 1.04:1 molar ratio, followed by calcination at 500 °C for 5 h and then at 850 °C for 12 h. ALD coatings were performed at 120 °C in a fluidized-bed reactor using trimethylaluminum (TMA) and either H<sub>2</sub>O or O<sub>3</sub> as reactants. Each cycle consisted of a 90 s TMA pulse, 120 s Ar purge, 90 s reactant pulse (H<sub>2</sub>O or O<sub>3</sub>), and 120 s Ar purge; four cycles ("4Al-H<sub>2</sub>O" and "4Al-O<sub>3</sub>") yielded an 5 nm conformal shell as confirmed by TEM. Coin cells were assembled with the coated cathodes, lithium metal anodes, a polypropylene separator, and 1 M LiPF<sub>6</sub> in EMC:EC:DMC (1:1:1 by volume). Electrochemical testing at 19 °C included galvanostatic cycling at 1 C, GITT for diffusivity calculations (per Chen et al.), and EIS over 100 kHz–0.1 Hz to extract  $R_{\rm ct}$  (following Bae et al.). All measurements were conducted in triplicate and statistically analyzed using one-way ANOVA with Tukey's post hoc test.

#### 5 Conclusion

Summary of main contributions: We systematically tuned the Li:Al atomic ratio in O<sub>3</sub>-based ALD supercycles (0.05–0.20:1) on NCM83 cathodes, applied GITT and EIS to quantify Li<sup>+</sup> diffusivity in the resulting 5 nm Al<sub>2</sub>O<sub>3</sub> shells, and identified an optimal Li:Al ratio (0.15:1) that minimizes grain-boundary and charge-transfer resistances while maximizing  $D_{\text{Li}^+}$ .

**Broader implications:** Our findings demonstrate that controlled Li incorporation during ALD not only preserves the protective nature of amorphous alumina coatings but also actively enhances ionic transport, offering a scalable route to improve high-voltage cathode performance and longevity.

**Limitations:** This study is limited by a fixed shell thickness (200 supercycles), a discrete set of Li:Al ratios, and predominantly ex situ characterization; real-time interfacial dynamics and long-term cycling effects remain unexplored.

Future research directions: We recommend expanding the ratio range and shell thickness, implementing in situ electrochemical spectroscopy during cycling, integrating atomic-scale modelling of Li pathways, and evaluating full-cell performance under practical operating conditions.

#### References

- [1] [Reference 1]
- [2] [Reference 2]
- [3] [Reference 3]
- [4] [Reference 4]
- [5] [Reference 5]
- [6] [Reference 6]