

Lecture XX

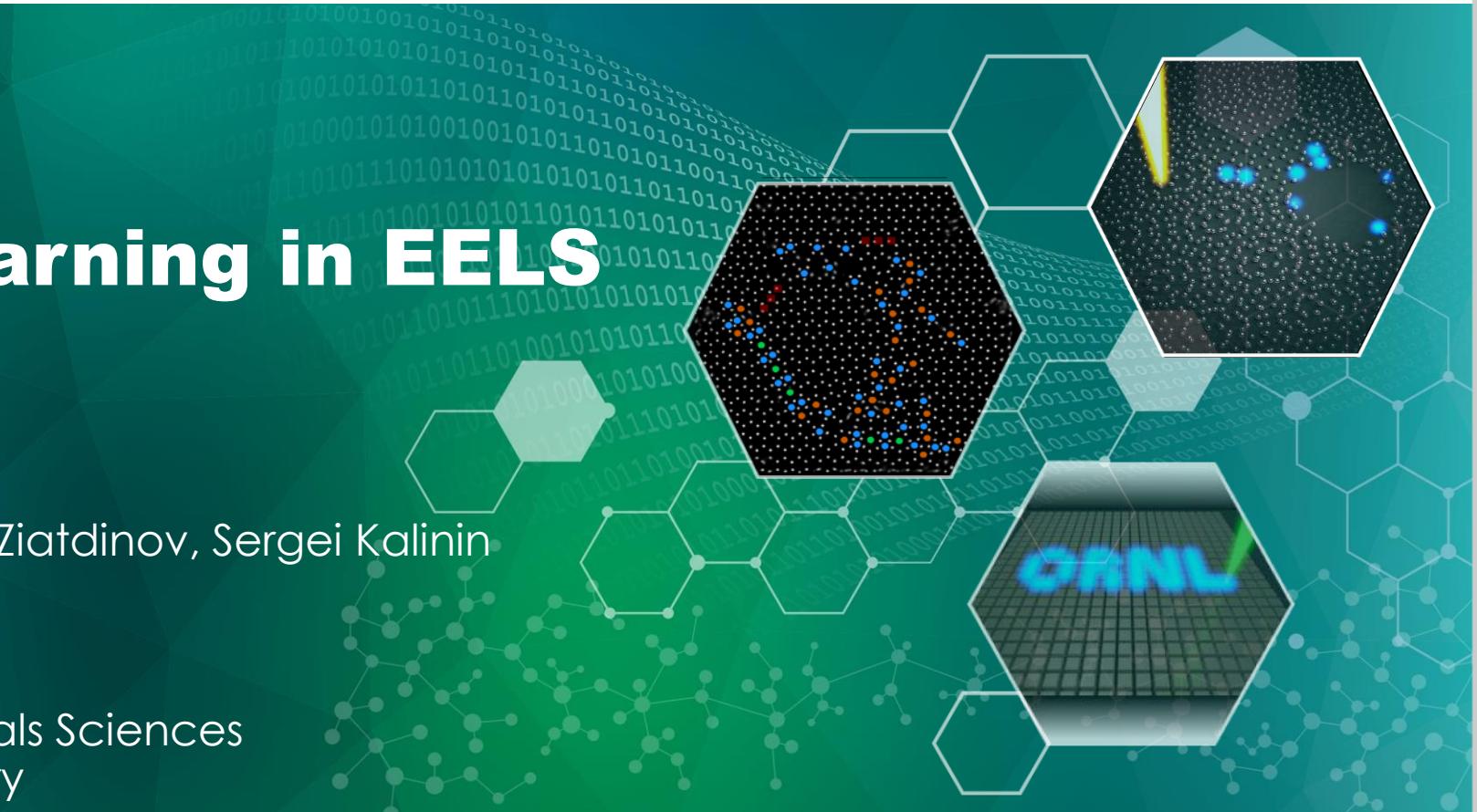
Deep kernel learning in EELS and 4D-STEM

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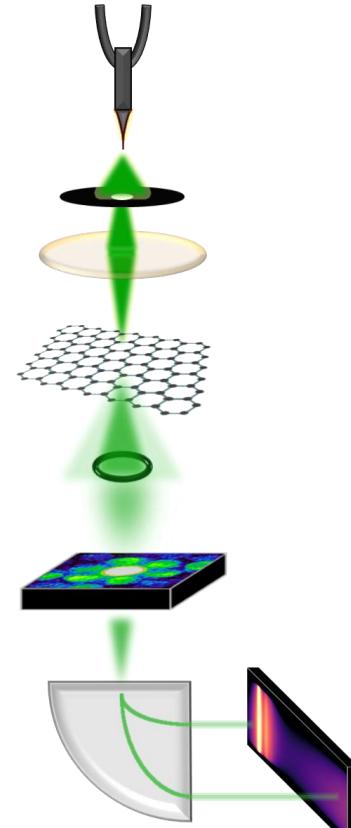
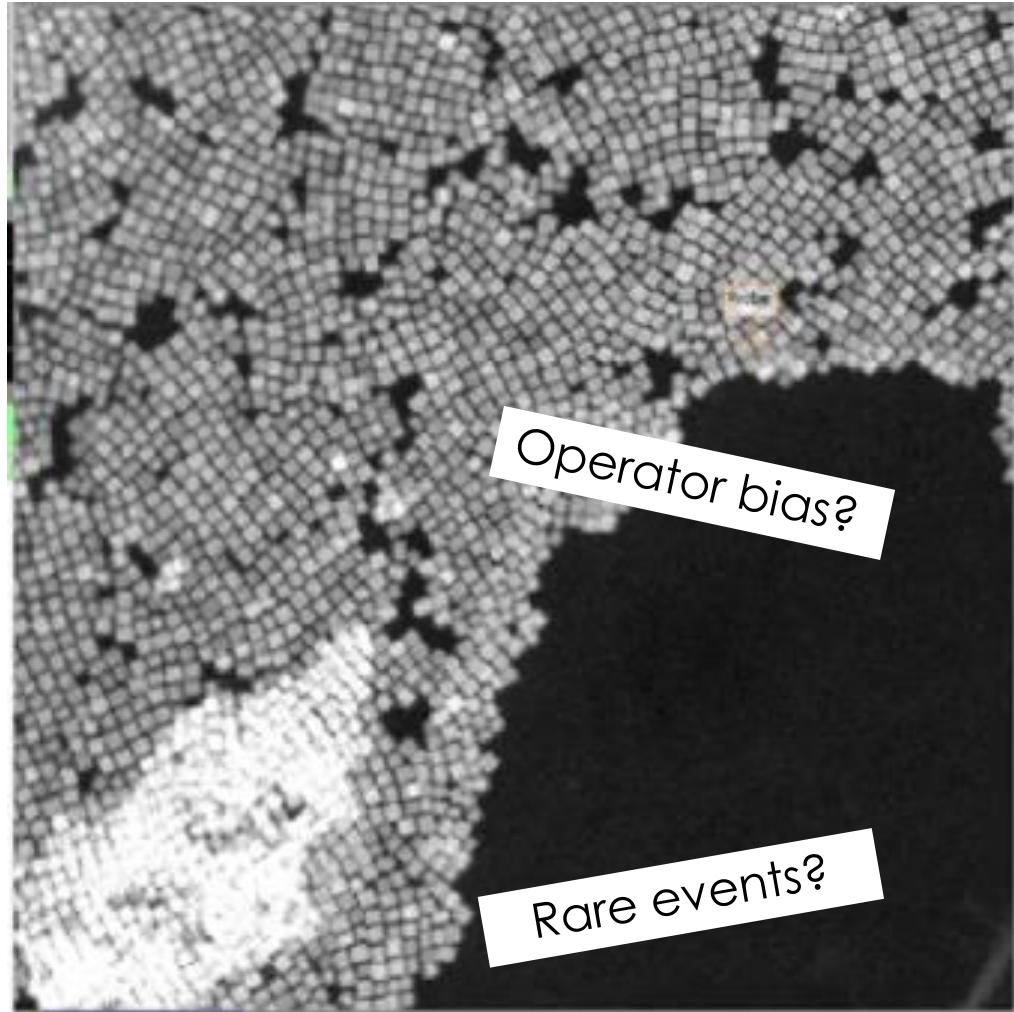
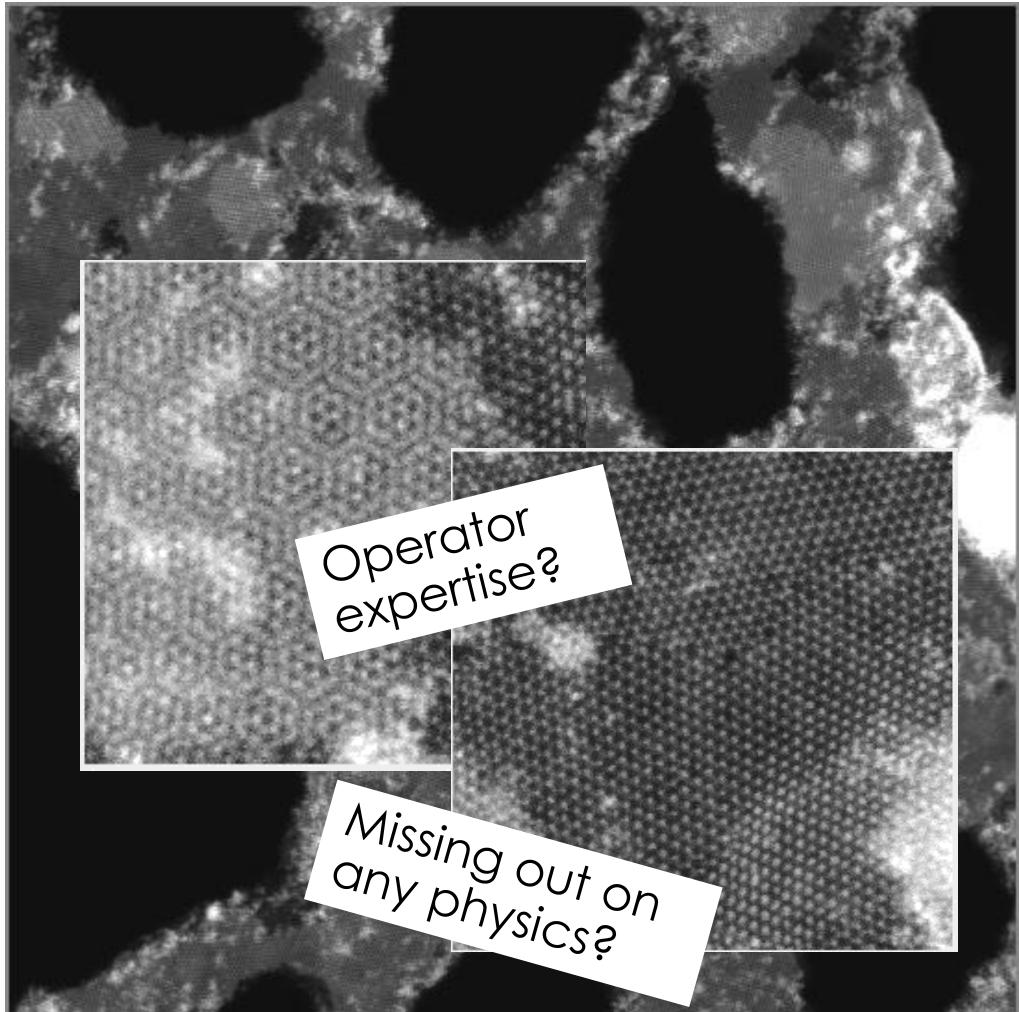
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Center for Nanophase Materials Sciences
Oak Ridge National Laboratory

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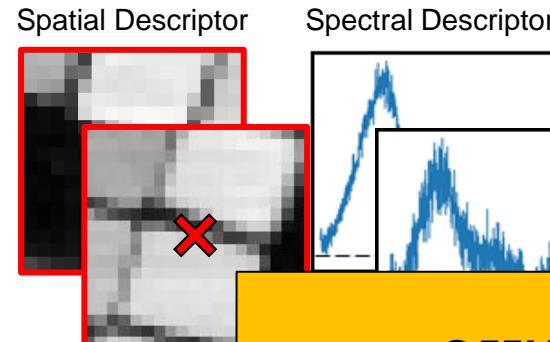
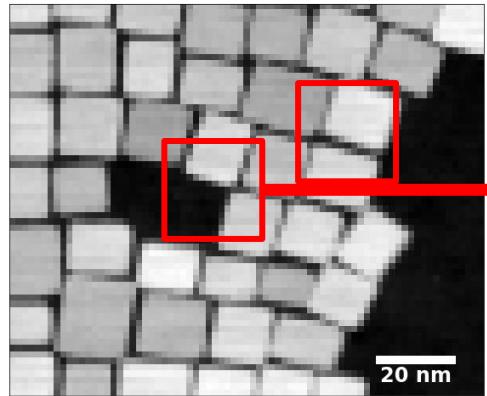


How do we understand structure property relationships



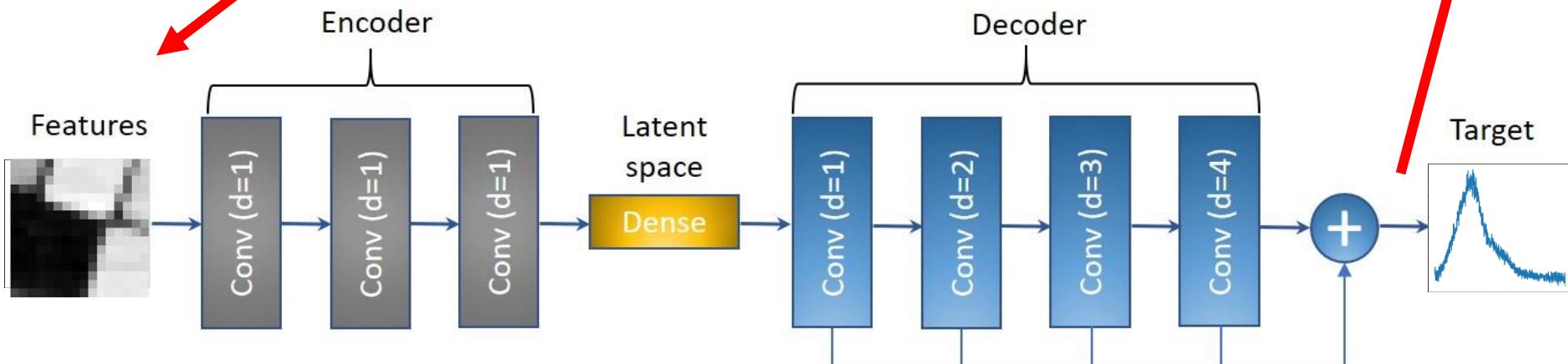
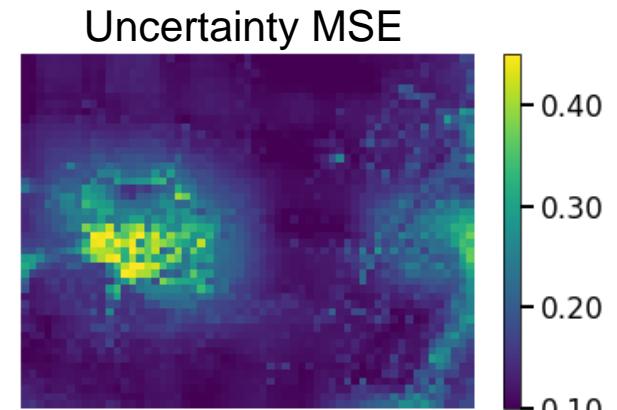
Q: Where should (would) you acquire analytical measurements? EELS spectrum image / 4D STEM, point spectra, where? **You can't say all points!**

Recall: Structure-property correlations: Autoencoder neural network



OFFLINE analysis!

Training can take some time.

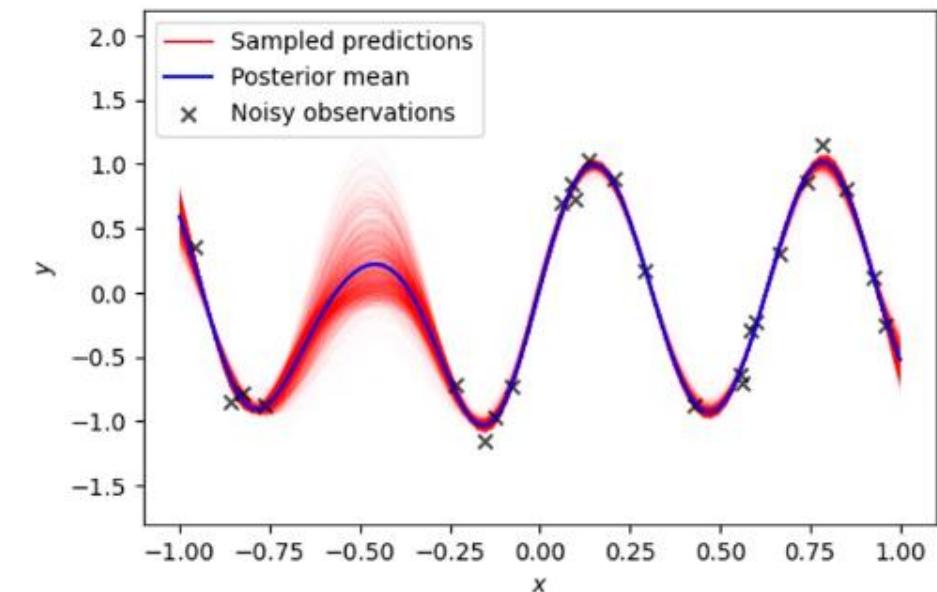


Automated Experiment

How to perform sparse measurements (intelligently)

in real time?

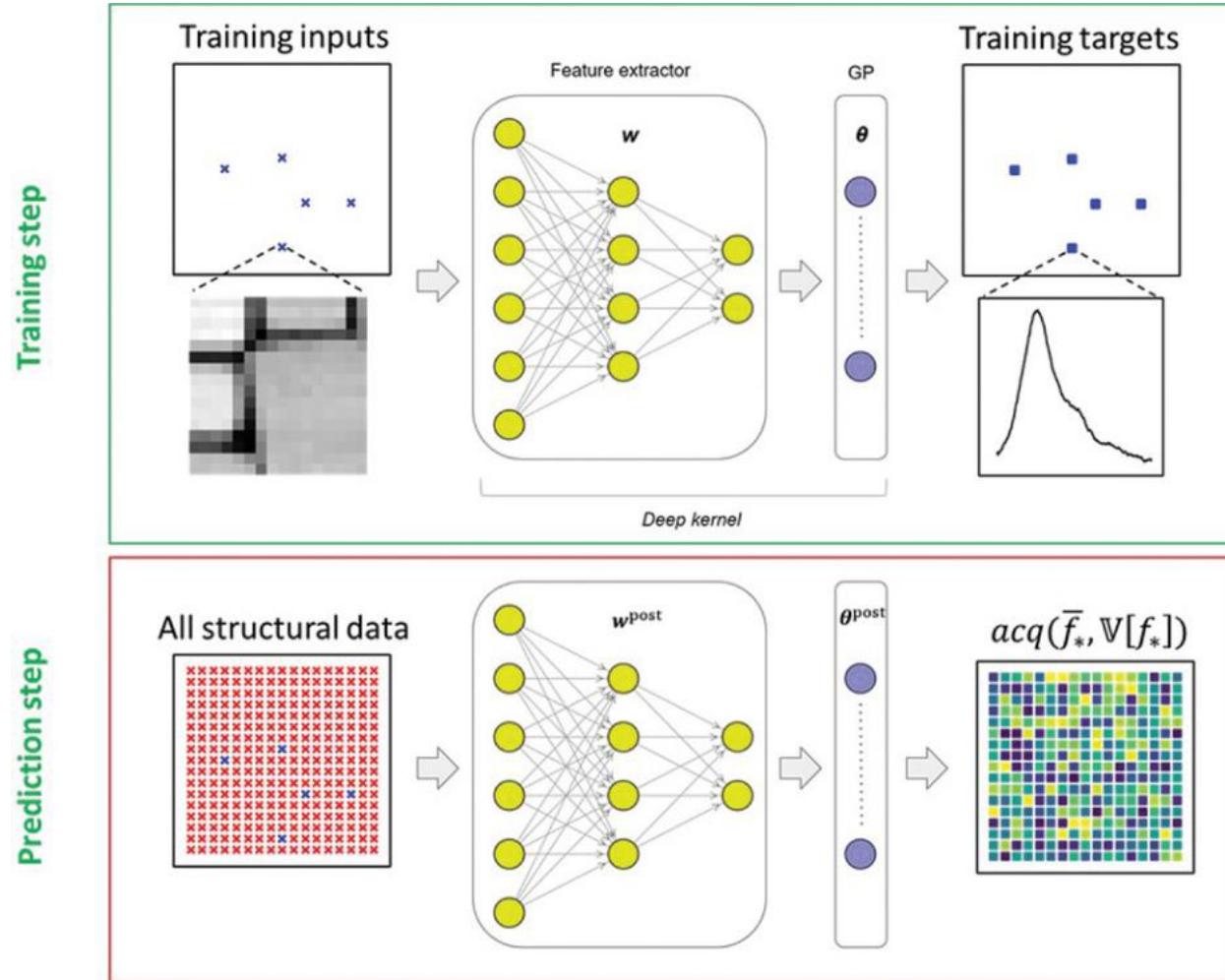
- Initial attempts →
 1. use gaussian process (**GP**) to “interpolate” the measurement space,
 2. then decide next measurements with Bayesian Optimization (**BO**)
- Gaussian Process → requires a kernel for reconstruction.
- - What kernel do we use?
 - Kernel length?
 - Periodic? Atoms... ok, lot of tuning... anything else?
 - Radial basis function (RBF) – maybe..
 - Custom kernel? But how to develop it?



If we could intelligently select a kernel... quickly!

See **lecture 14** for GP and BO details if you missed it

Deep kernel learning (DKL)



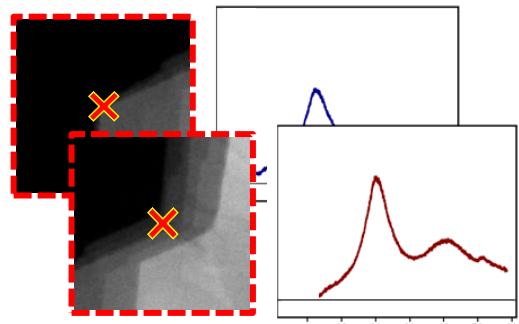
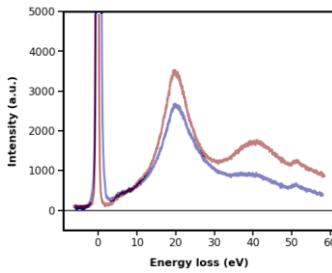
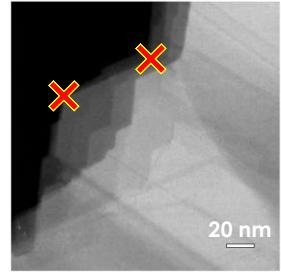
Tuning knobs

- “Scalarizer”
 - Physics enters here
- Acquisition function
 - How to navigate given the predicted space
 - Expected improvement
 - Upper confidence bound
 - Uncertainty-based
 - Thompson sampling

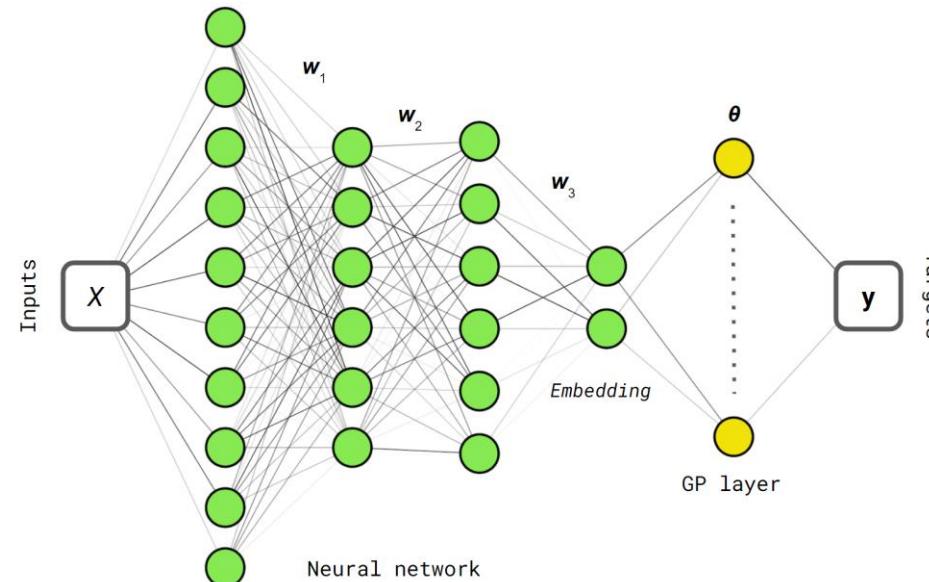
Physics-based feature engineering:

Deep kernel learning – Bayesian optimization

Specify physics criteria



Active learning



Acquire structural data

Measure a spectrum

Train DKL model with new data

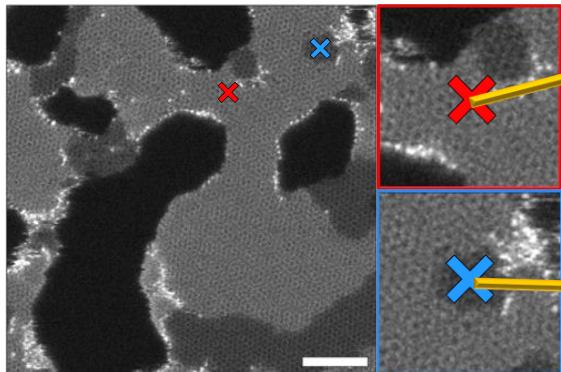
Decide next position (optimize physics criteria)

Allows navigation of the system to search for physics

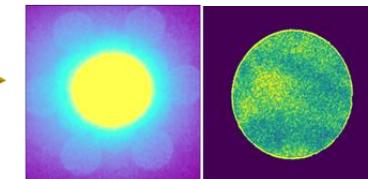
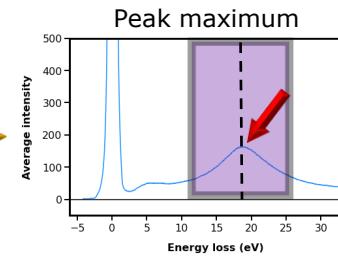
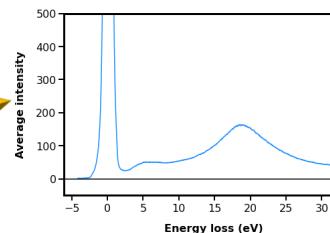
Actively learning structure-property relationships: Deep kernel learning (DKL)

No pretraining!

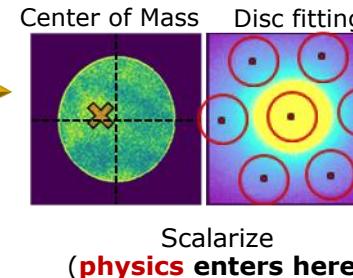
This is all on-the-fly



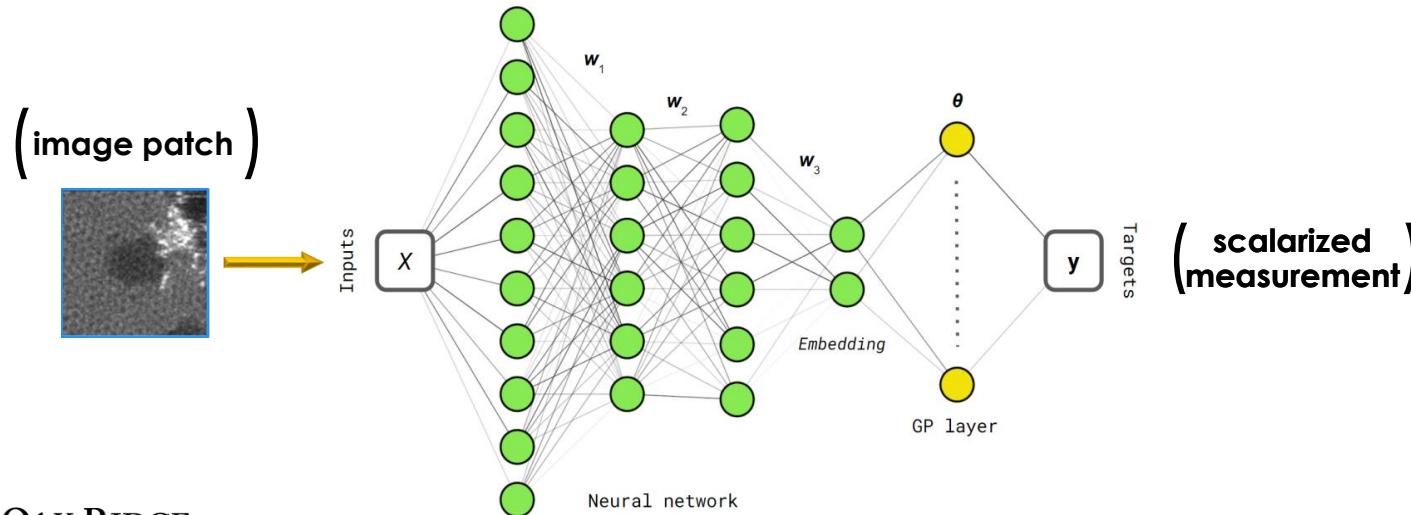
Complete
structural
image



Measurement
(1D or 2D)

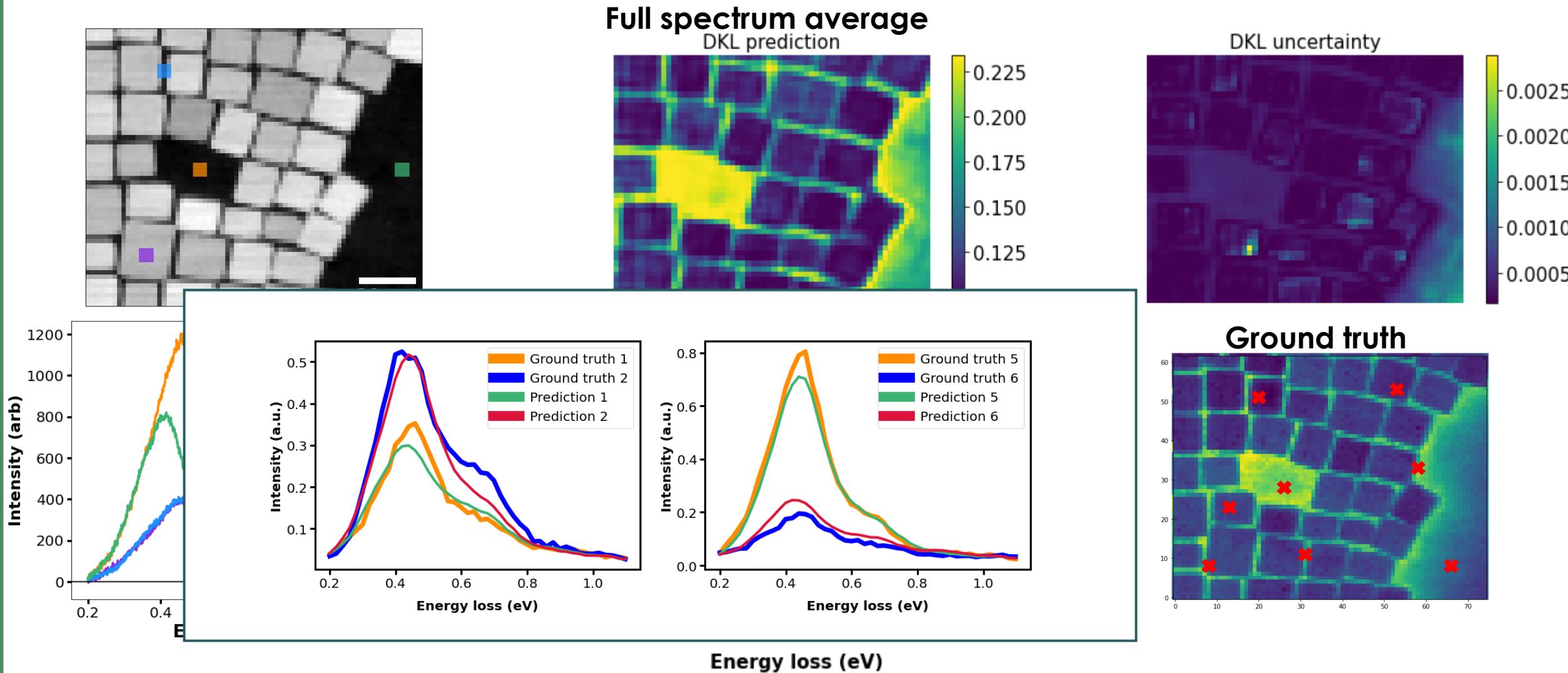


Scalarize
(physics enters here)



1. A complete **structural image**
2. Image **patches**
3. Perform a **measurement** @ (x_1, y_1)
4. “**Scalarize**” (**physics** incorporation)
5. Learn **correlation** between image signal and measurement signal (training step)
6. **Next measurement** @ (x_2, y_2) based on learned correlation
7. Repeat

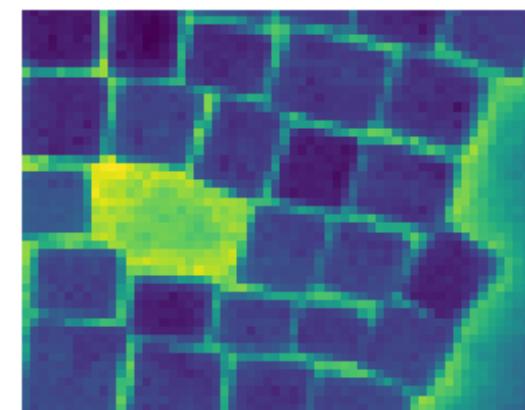
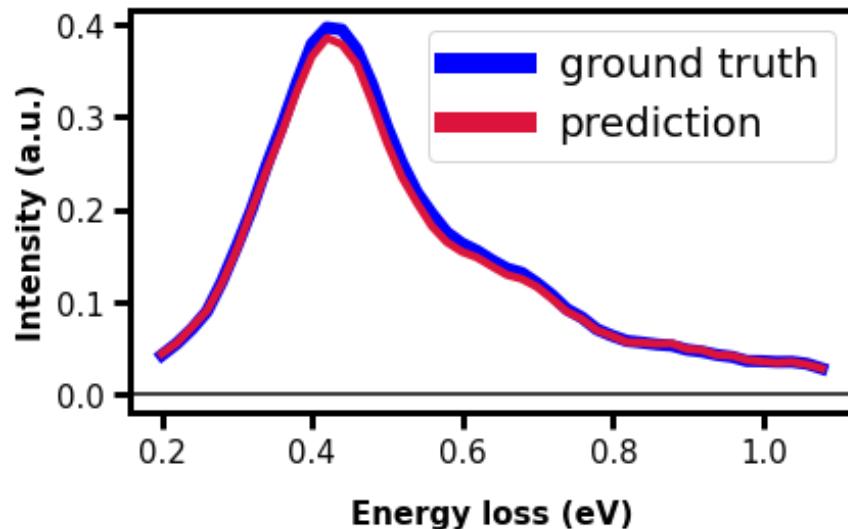
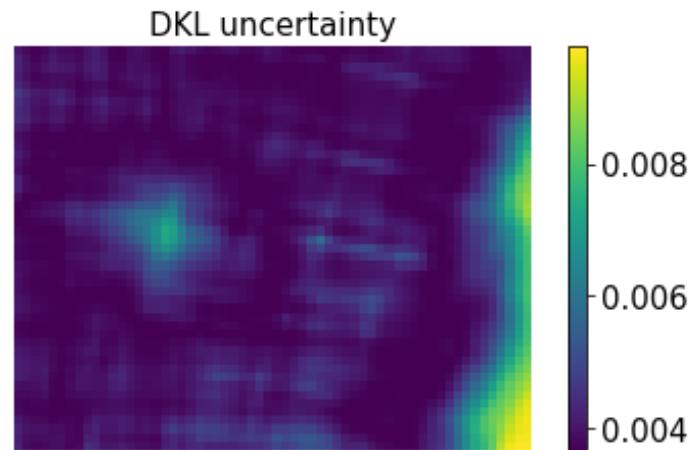
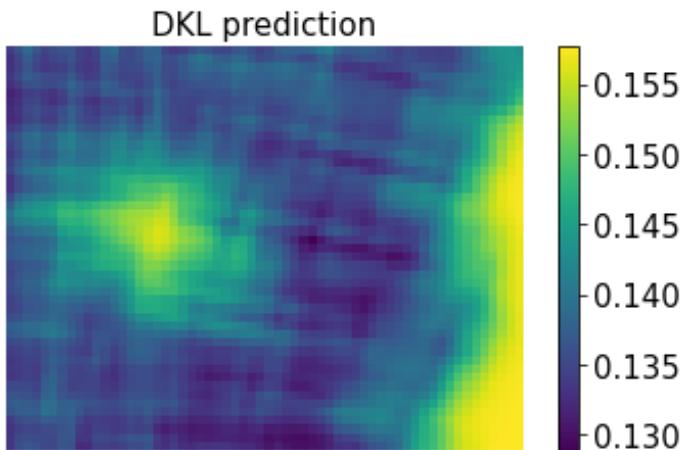
DKL with **FULL** pre-acquired data (no active learning or BO)



- This is actually “**vector**” DKL which **reconstructs entire 1D spectrum (or 2D image)**
- Is not so amenable for automated experiment, however

DKL with partial pre-acquired data (no active learning or BO)

1% random sampling



Active learning with DKL still on pre-acquired data using a scalarizer

Scalarizers

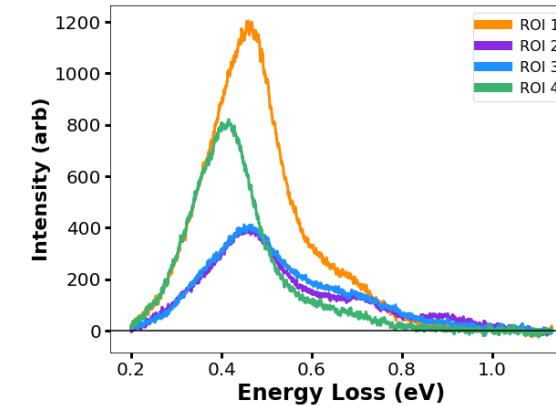
- Peak max in spectral range
- Ratio of peaks (e.g., L2,3 ratio, surface plasmon / bulk plasmon)
- Number of peaks
- Peak width
-
- Center of mass (magnitude, angle...)
- Charge density
- Strain (diffraction disc separation in fourier space)
-

Acquisition functions

- Expected improvement
 - Upper confidence bound
 - Uncertainty-based
 - Thompson sampling
- **Handles the next measurement choice**, based on the currently learned structure-property relationship (i.e., prediction)
- “exploration” vs “exploitation”

Active learning with DKL still on pre-acquired data using a scalarizer

```
1 def scalarize_acq(obj: np.ndarray) -> np.ndarray:  
2     """  
3         Scalarize acquisition function (n_samples x vector_size)  
4         by averaging over a selected energy band.  
5     """  
6     band = [0, -1]  
7     obj = obj[:, band[0]:band[1]].mean(-1)  
8     return obj
```



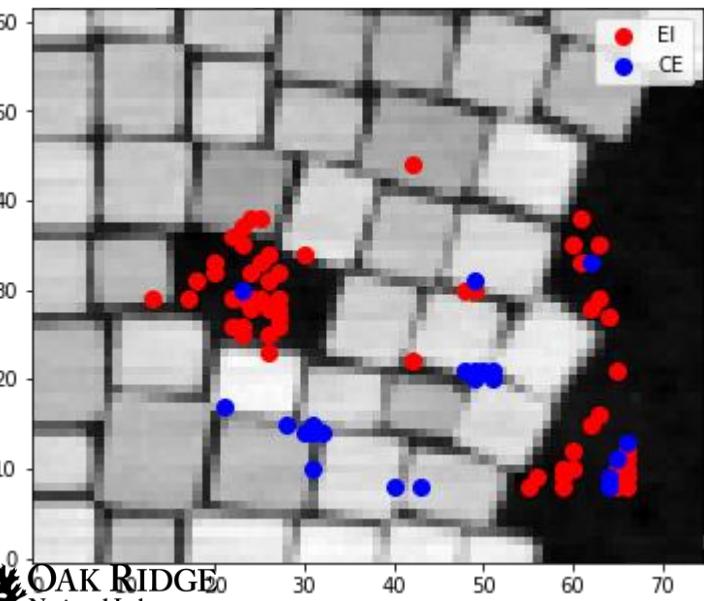
Measure & scalarize



Train

New measurement
based on A.F.. Repeat

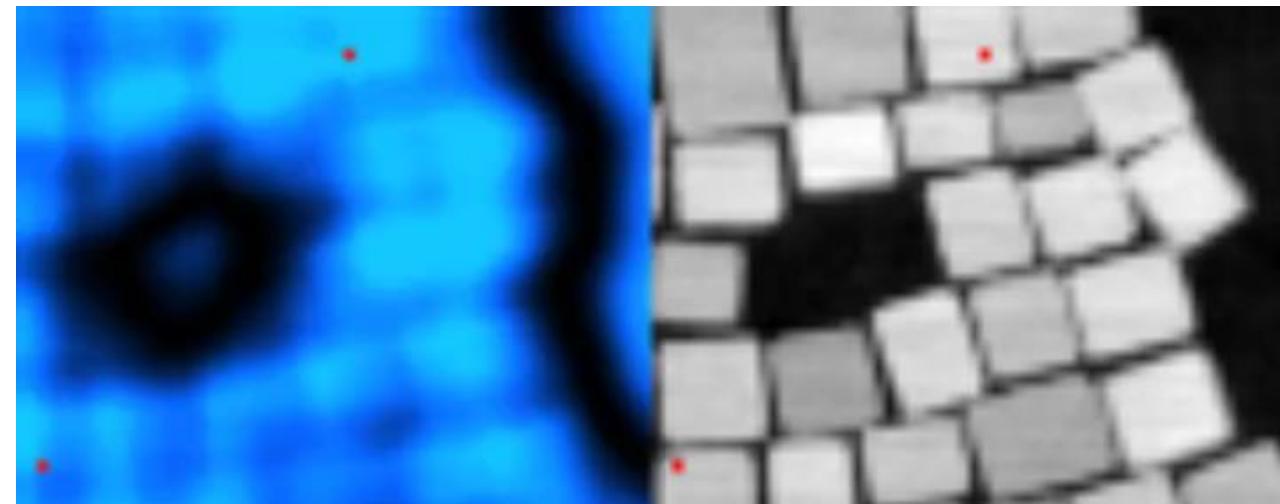
Next measurement point: *argmax()*



Choice of
**acquisition
function** affects
pathway

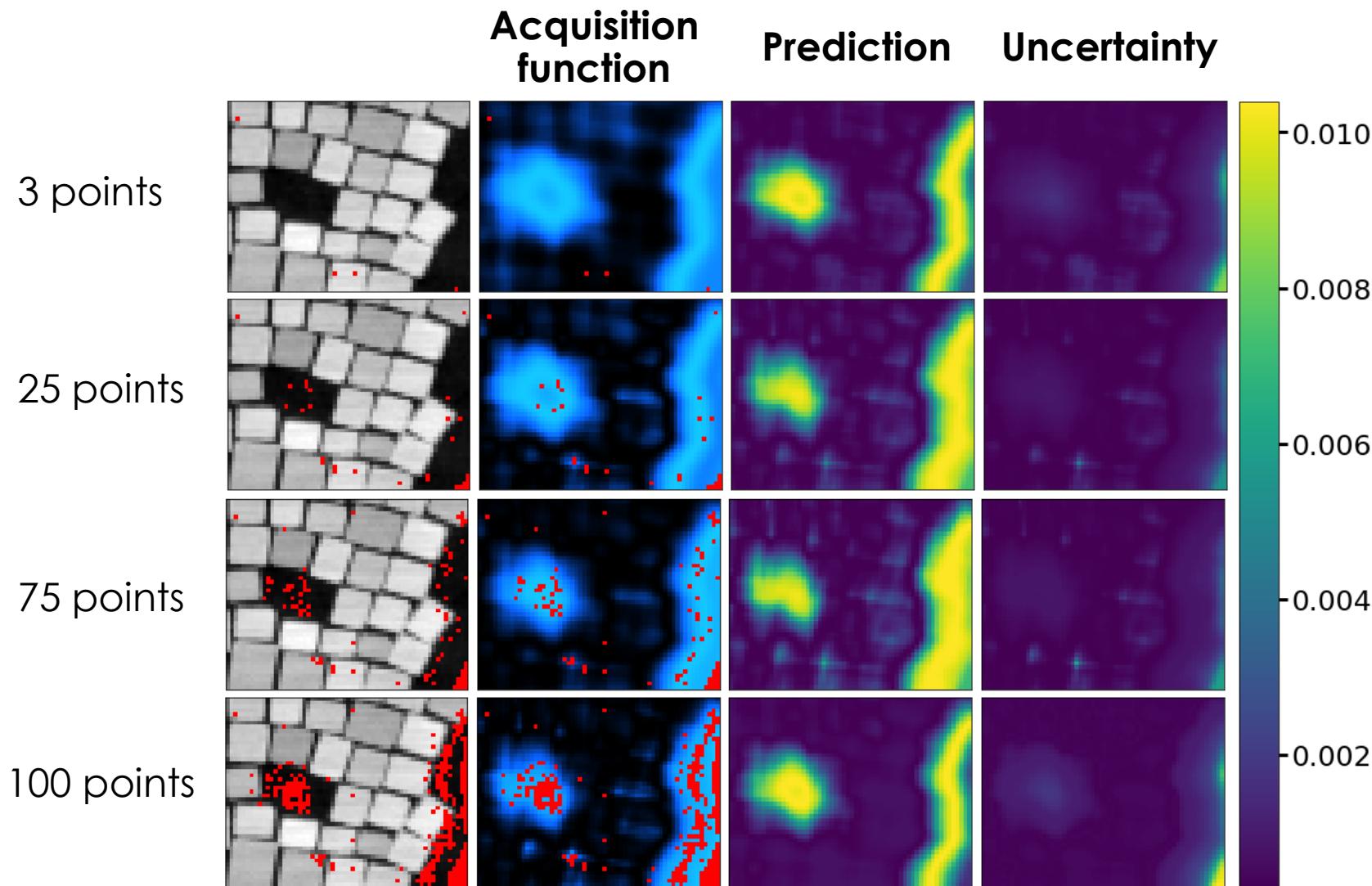
- Averaging whole (background subtracted) spectrum → **favoring the strong dipole plasmon**

Acquisition function



Evolution and performance in time

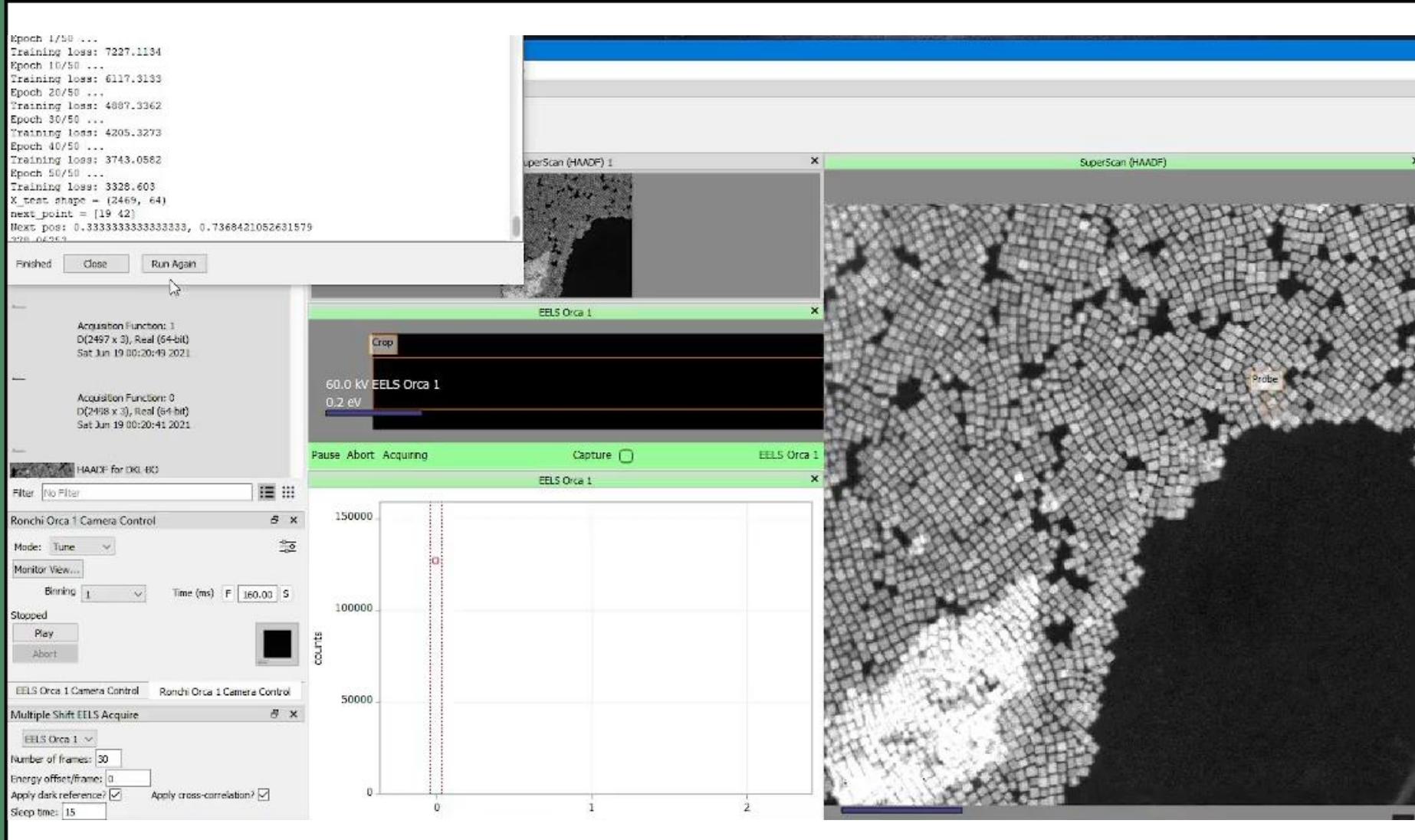
Have a look at predictions



- A good approximate structure-property relationship is **learned very quickly**
- In practice, we do not need to train at every new measurement
- Alternatively, can halt training after a few measurements (or some criterion met), then follow the acquisition function map
- Remember, **this is a prediction of the scalarized value** not the full spectrum

A real automated experiment

Can ML run experiment as a scientist?



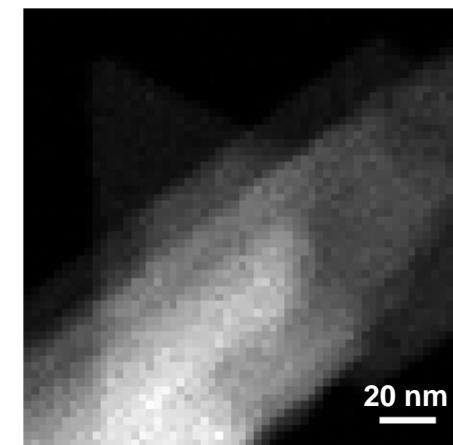
With **access to microscope controls** (Python API), this is deployed on an instrument

- **Continuously learning** structure-property relationships with each new measurement
- **Dose reduction!**
- In some sense, this is sparse sampling, but each subsequent point depends on the previous

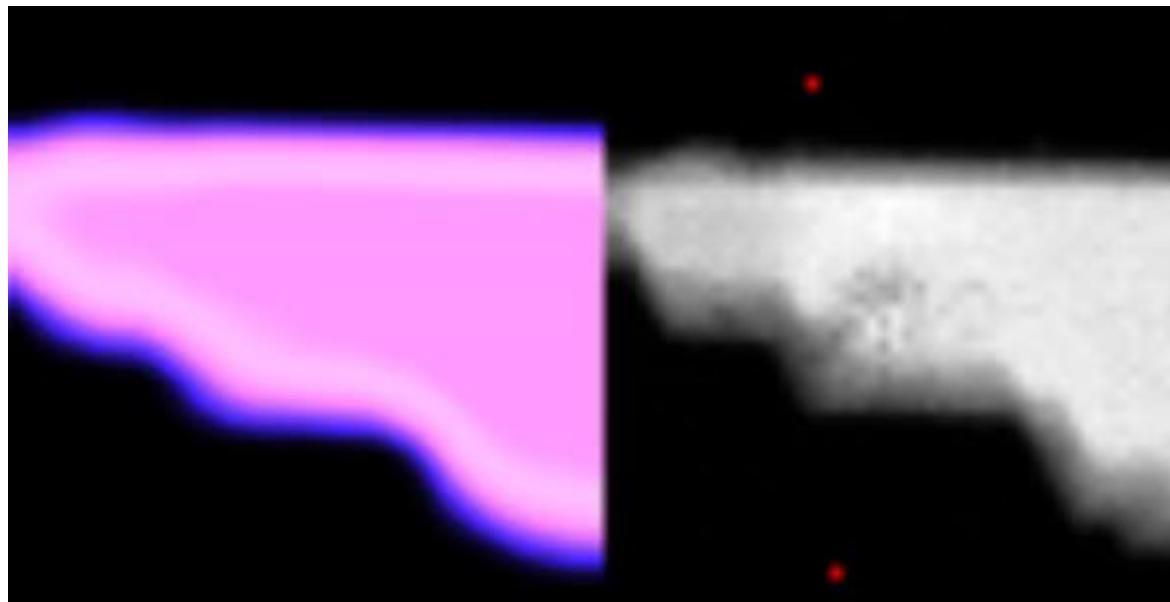
Beam is blanked during training steps

Physics-based feature engineering: MnPS_3

- Discovering physics in a “new” material MnPS_3



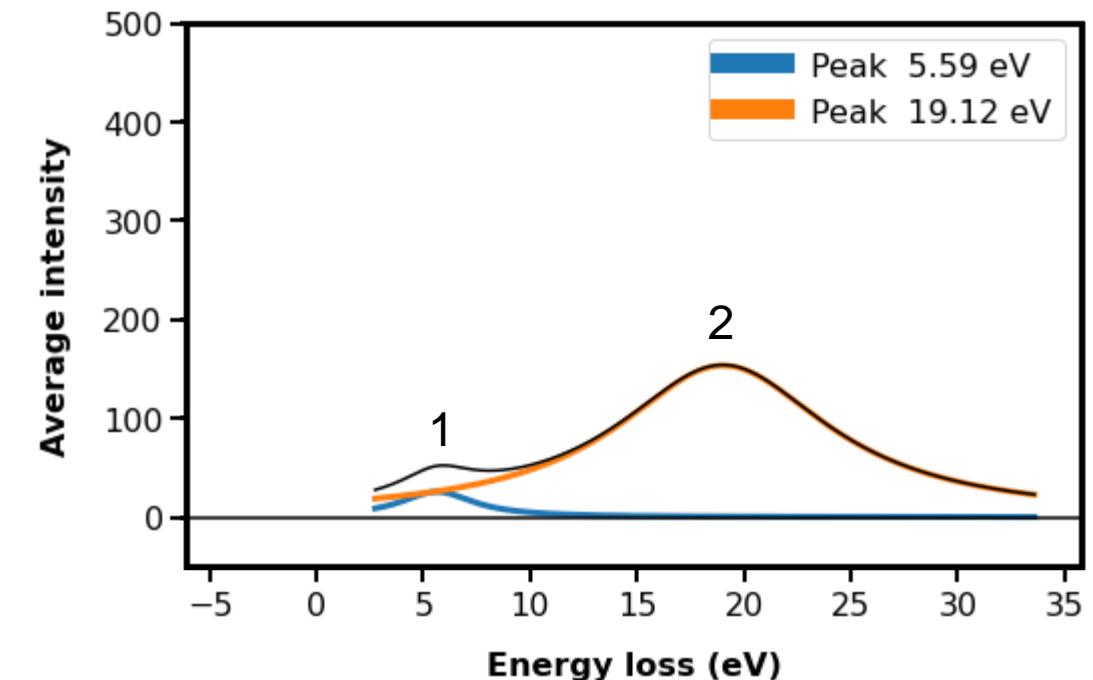
“Acquisition function”



HAADF-STEM
+ points visited

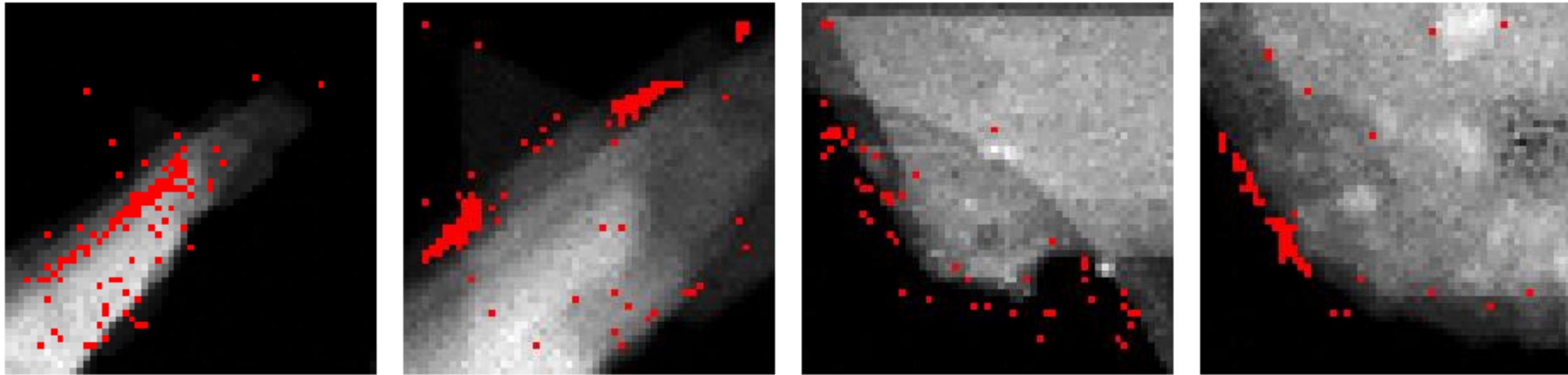
Physics search criteria:

$$\textit{Ratio} = \textit{Peak 1} / \textit{peak 2}$$



Reproducible

- Very similar behavior when searching for the same criteria elsewhere!
- Success!

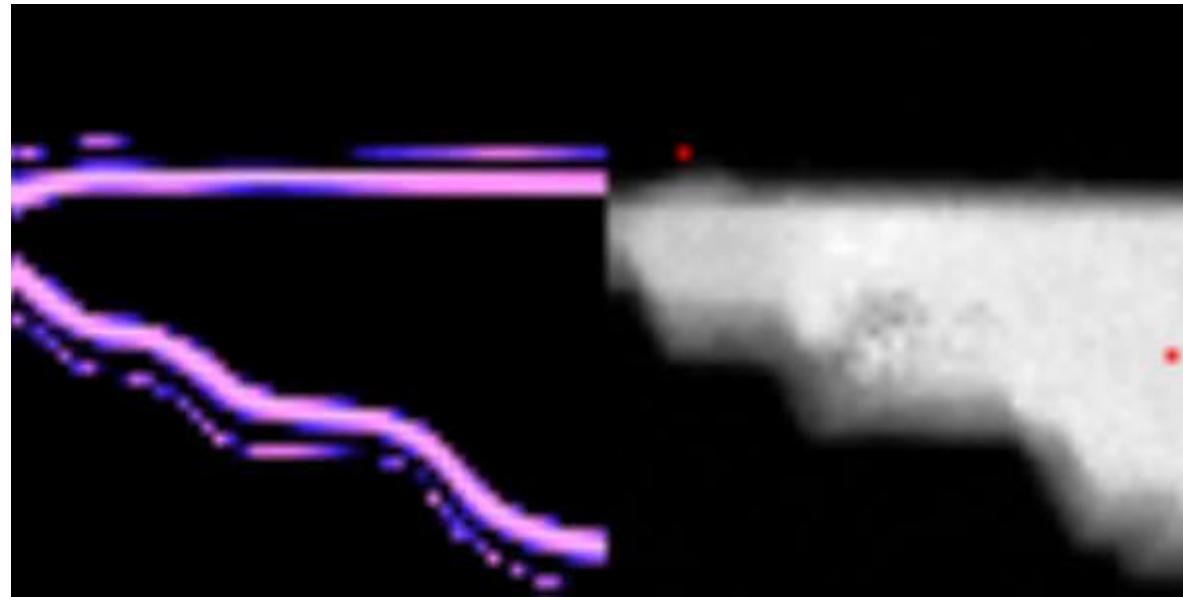


Discovery pathway depends on the reward structure (scalarizer that defines signature of physics we want to discover)!

Change the scalarizer... change the physics

- (Same region) **Simple physics search:** peak max in selected region

“Acquisition function”

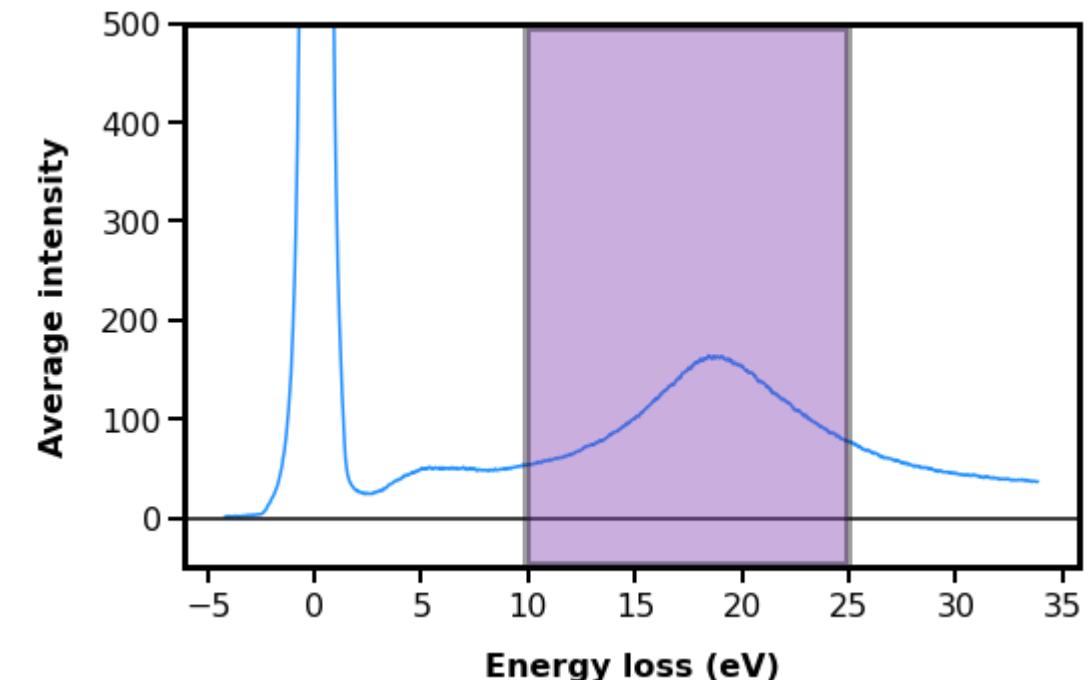


HAADF-STEM
+ points visited

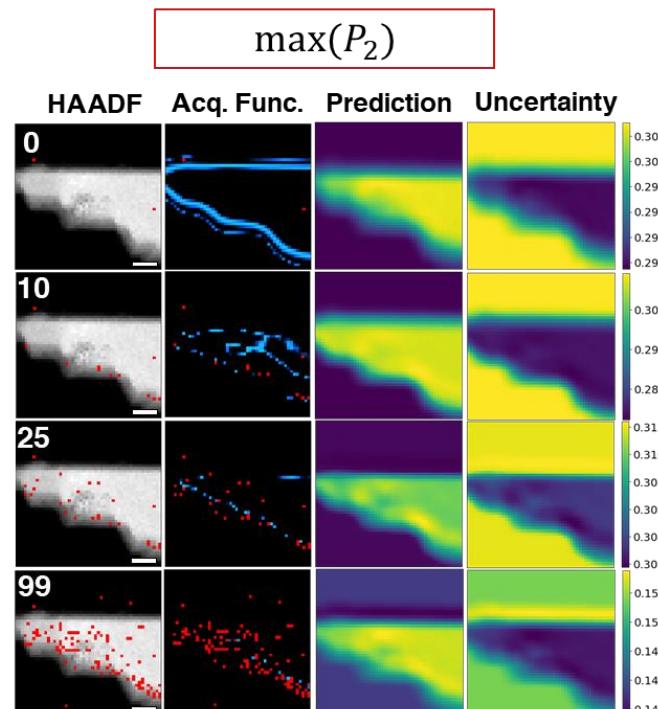
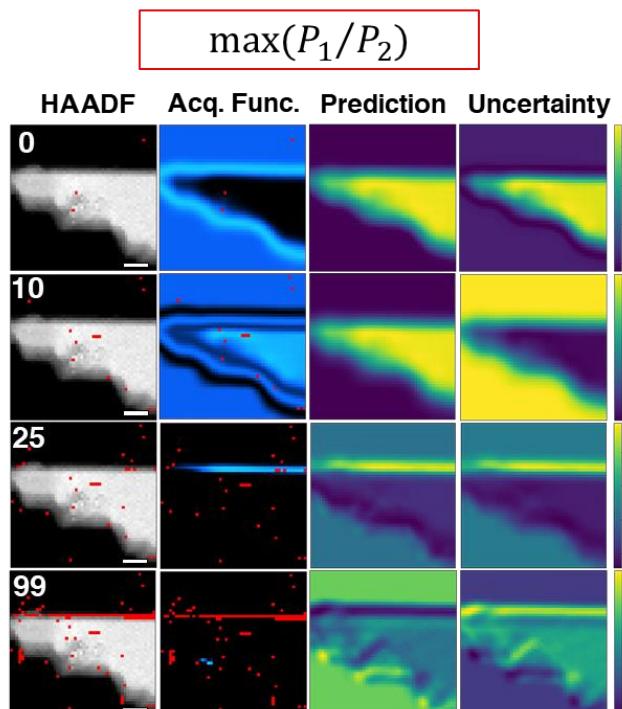
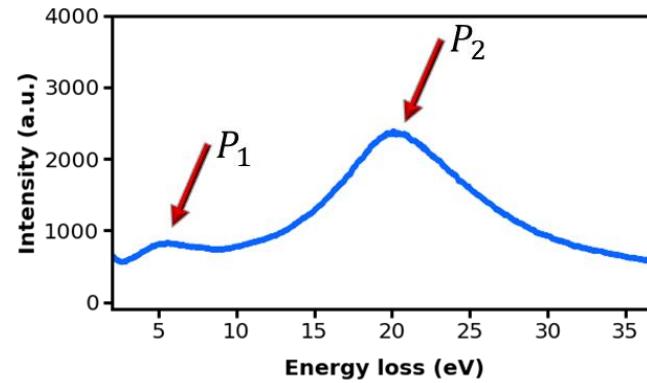
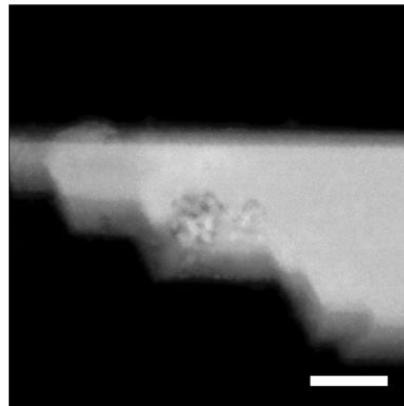
Physics search criteria:

$$\text{Maximize}(f)$$

(Specific peak intensity)

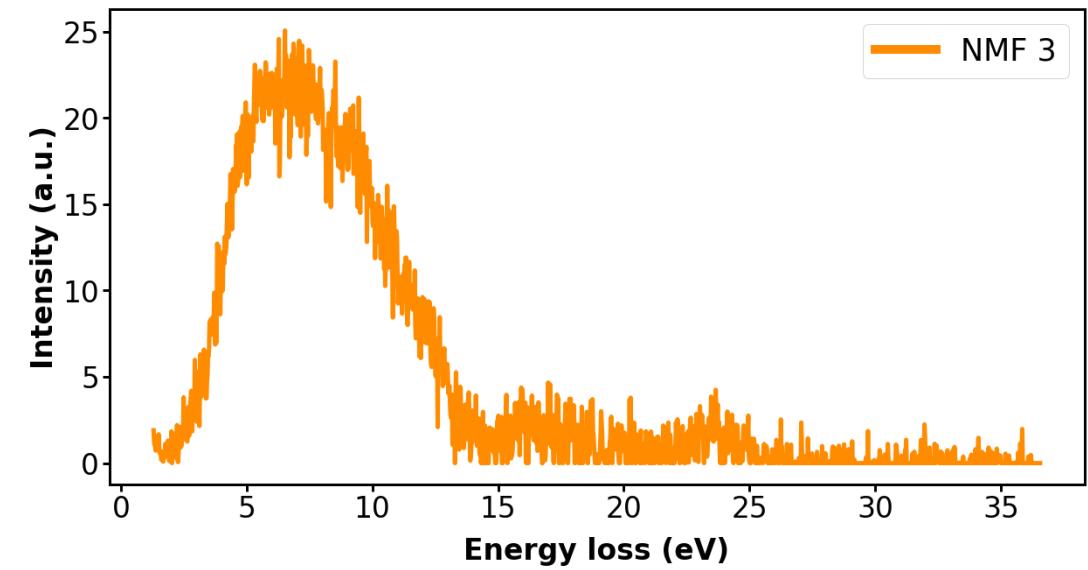
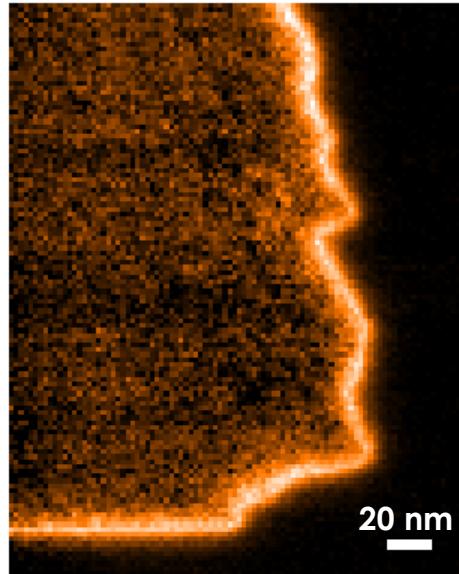


Summarize these automated experiments..

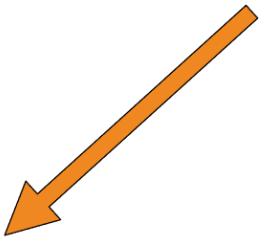


Confirmation

- Full hyperspectral EELS set collected and decomposed into few number of components (NMF)
- Indeed, there is edge plasmon activity near the energies used in the DKL

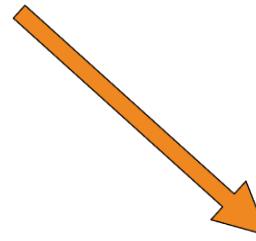


4D STEM



Center of mass

- Electric field
- Potential
- charge density

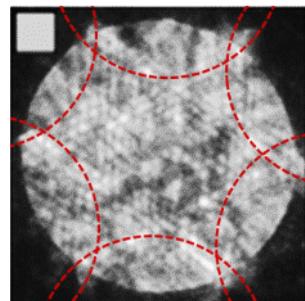
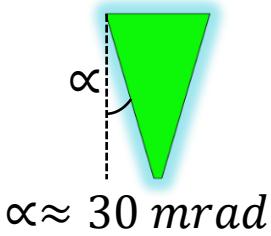


NBED disc fitting

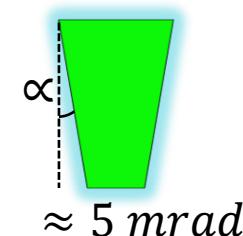
- Strain
- Grains

Require different probe conditions

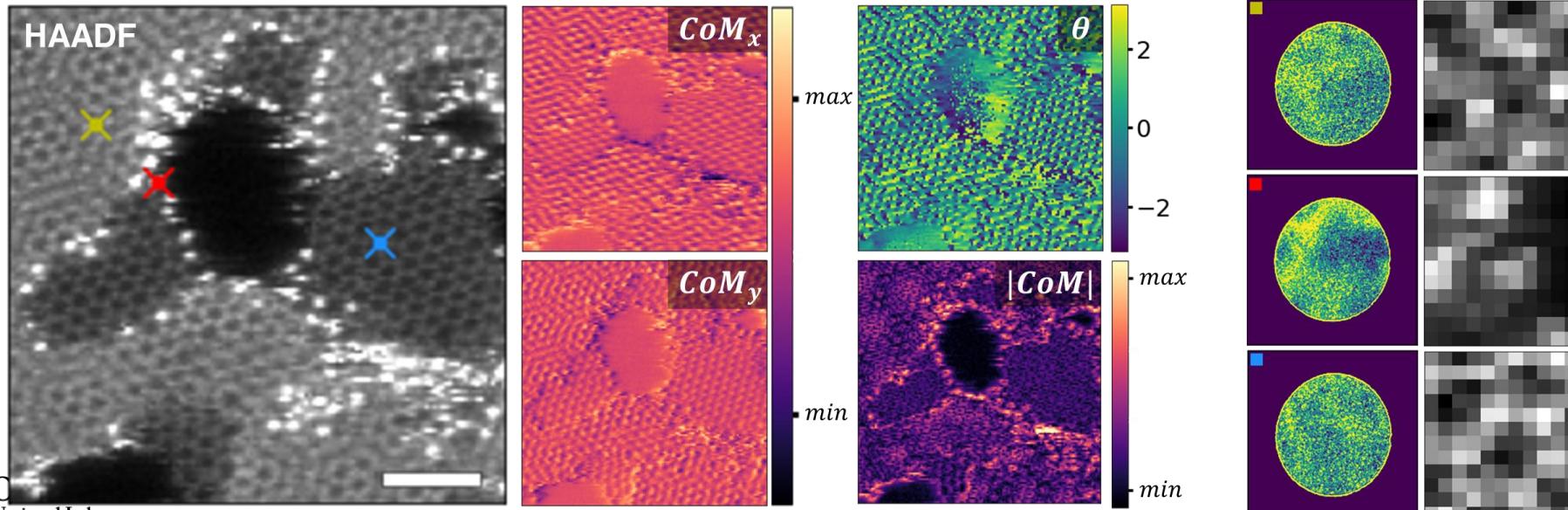
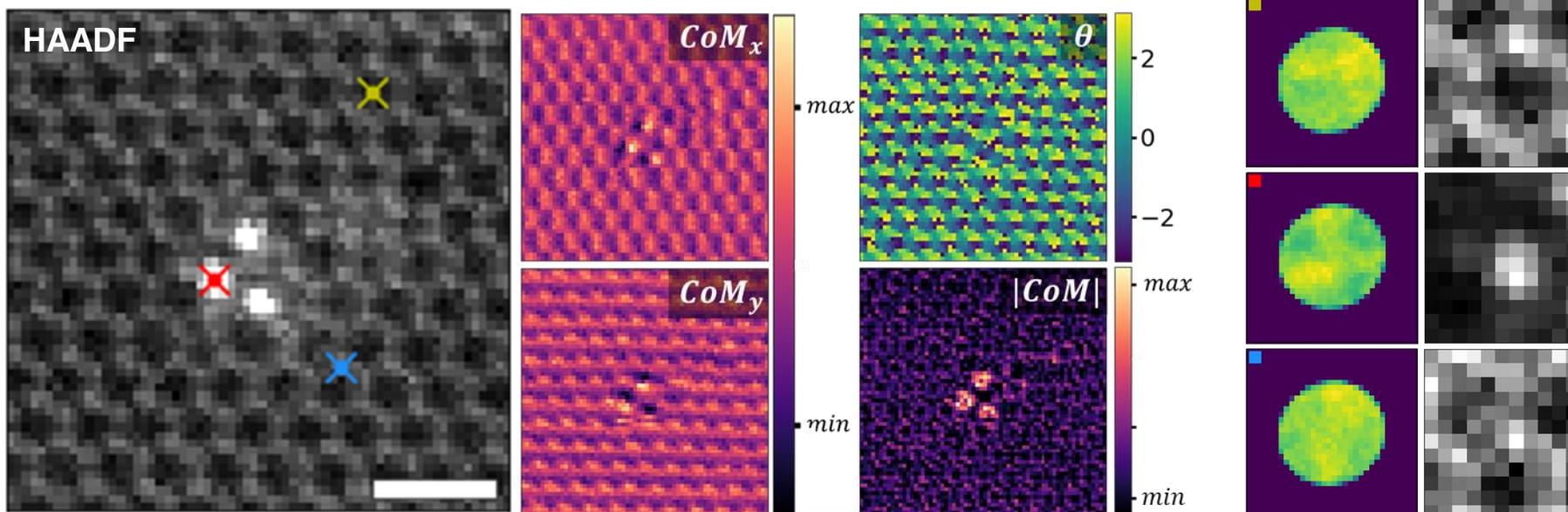
Overlapping diffraction discs



Non-overlapping diffraction discs



DKL: 4D STEM – center of mass (DPC)

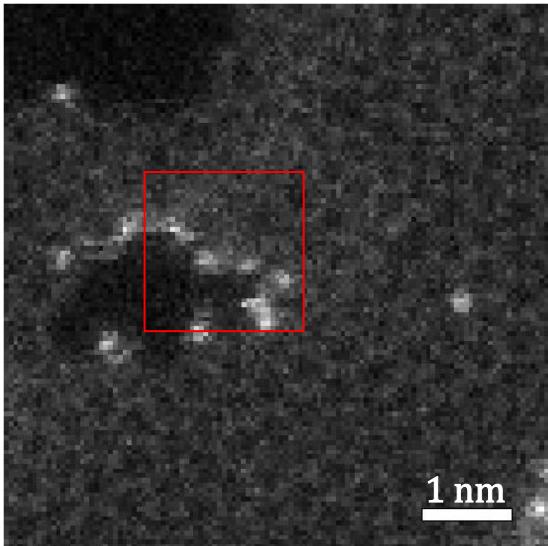


DKL: 4D STEM – center of mass

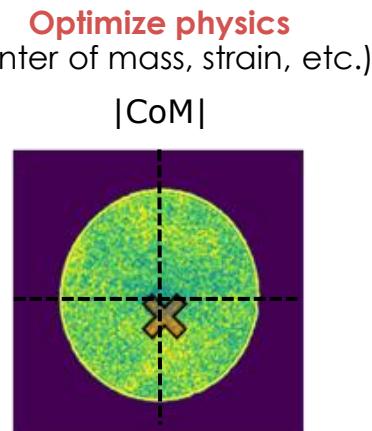
Example experiment:

- search for **maximum sample electric field**
- calculate $|\text{center of mass}|$ of diffraction pattern

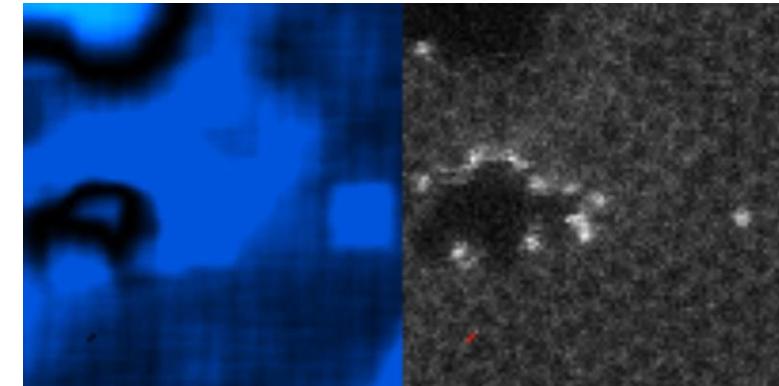
Full structural image



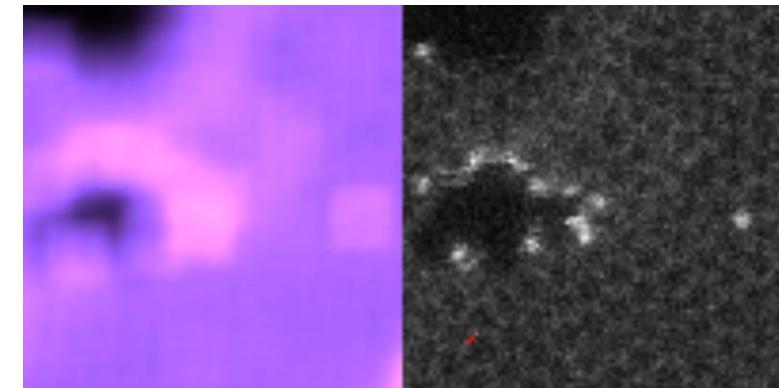
Measured quantity



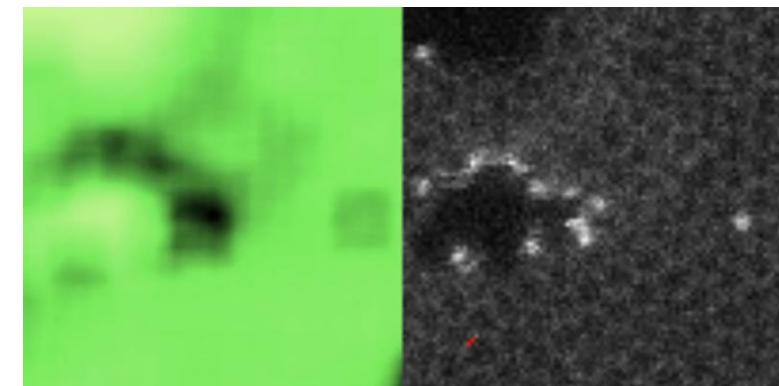
Acquisition function



Prediction map



Uncertainty map

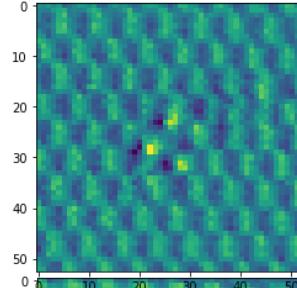


DKL: 4D STEM (pre-acquired data)

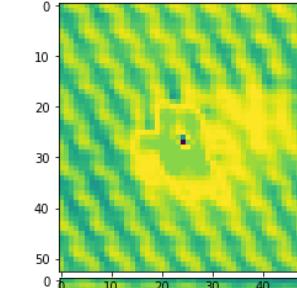
- Choosing **different scalarizers**, does one perform “better”?
- Some interesting (real) artifacts appear in some predictions

CoM_x

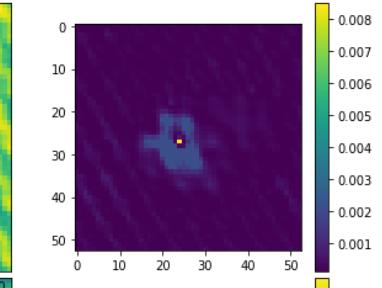
Ground truth



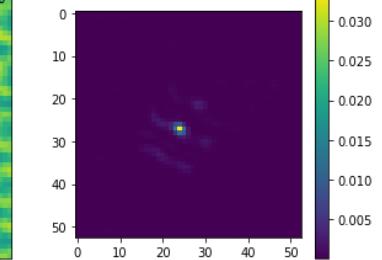
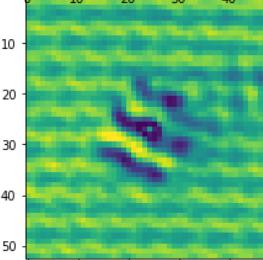
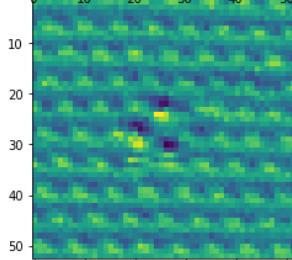
Prediction



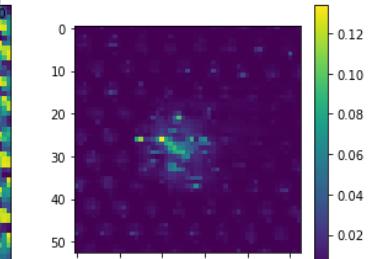
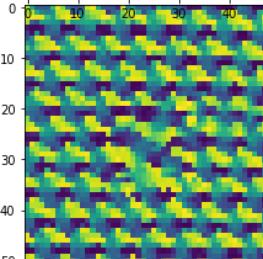
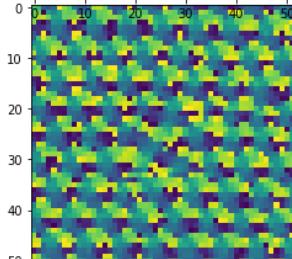
Uncertainty



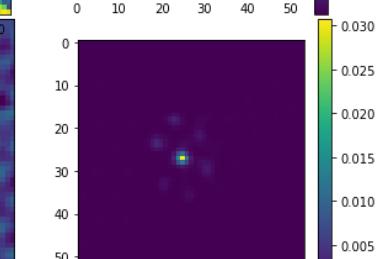
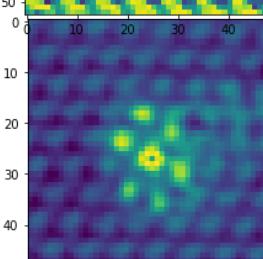
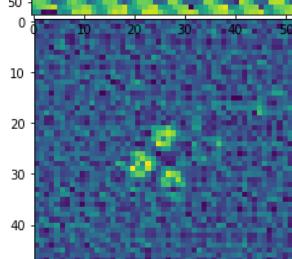
CoM_y



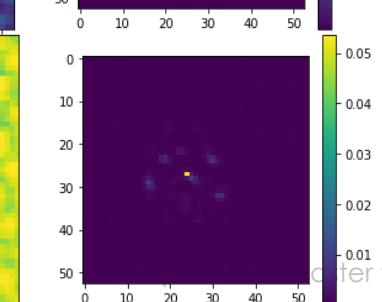
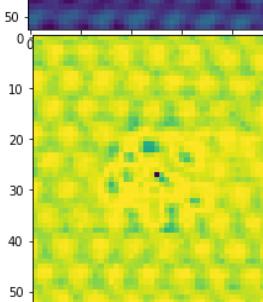
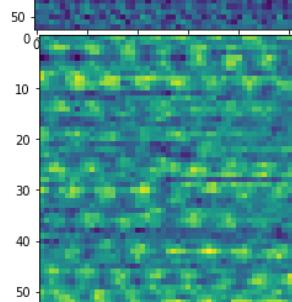
CoM_θ



$|\text{CoM}|$



Virtual
ABF



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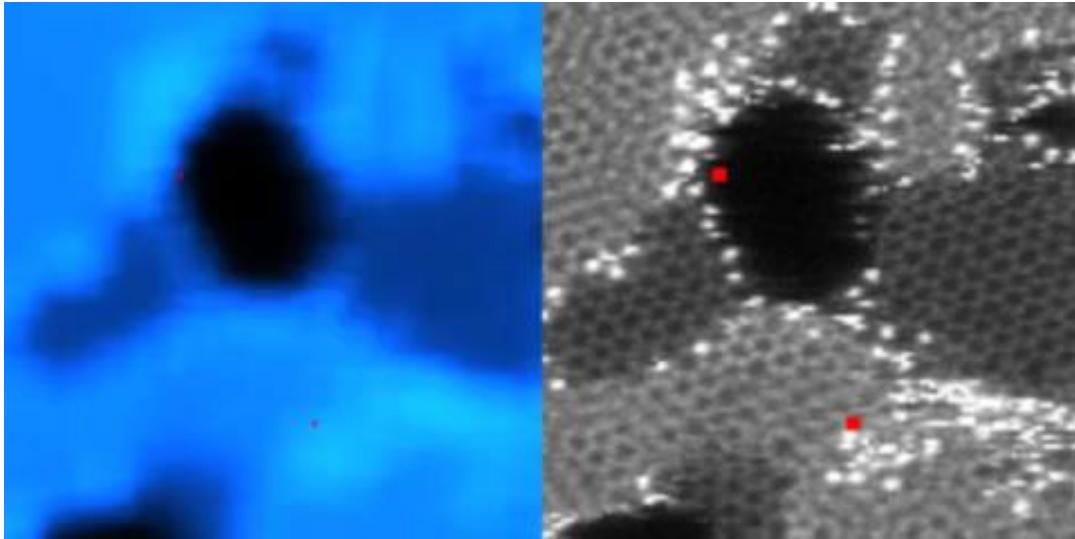
0.005

center to edit

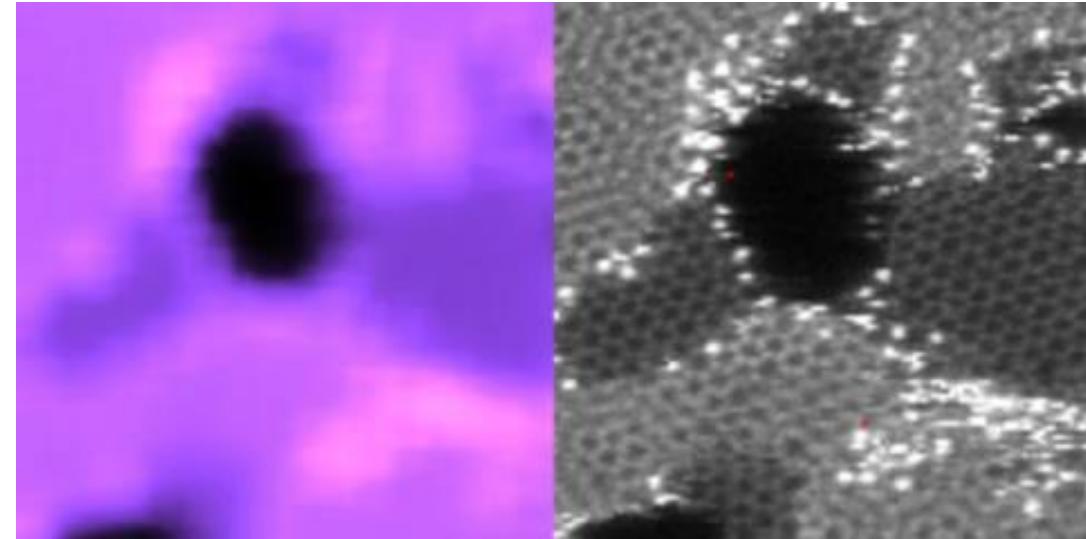
DKL: 4D STEM – active learning (pre-acquired data)

Sample: TBG (twisted graphene)
Scalarizer: DPC |**CoM**|

Acquisition function

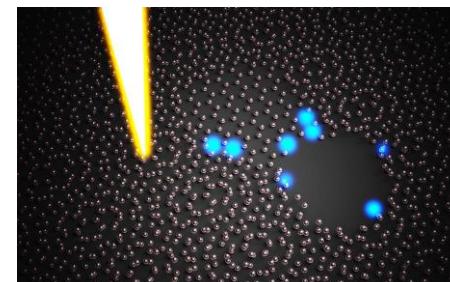


Prediction



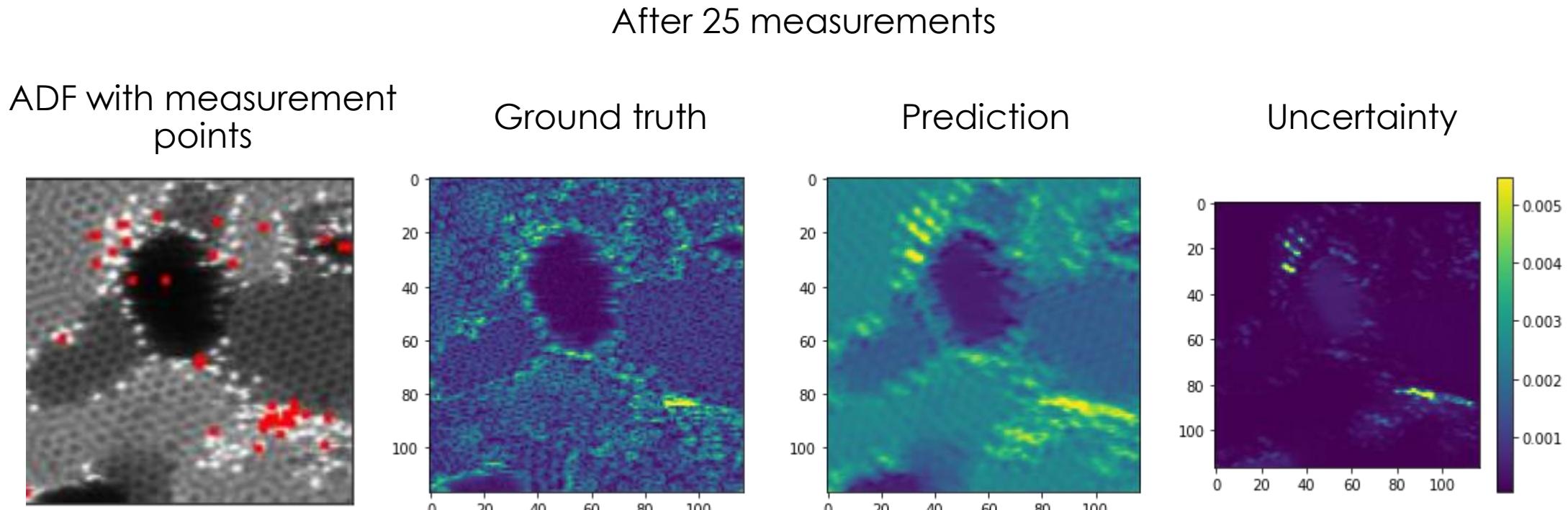
Next measurement position is based on *argmax*(Acq. Func.)

- Different **acquisition functions** can be used:
 - Expected Improvement (**EI**) (usually what was used)
 - Upper Confidence Bound (**UCB**), etc.
- Usually based on some combination of **prediction** and **uncertainty**.

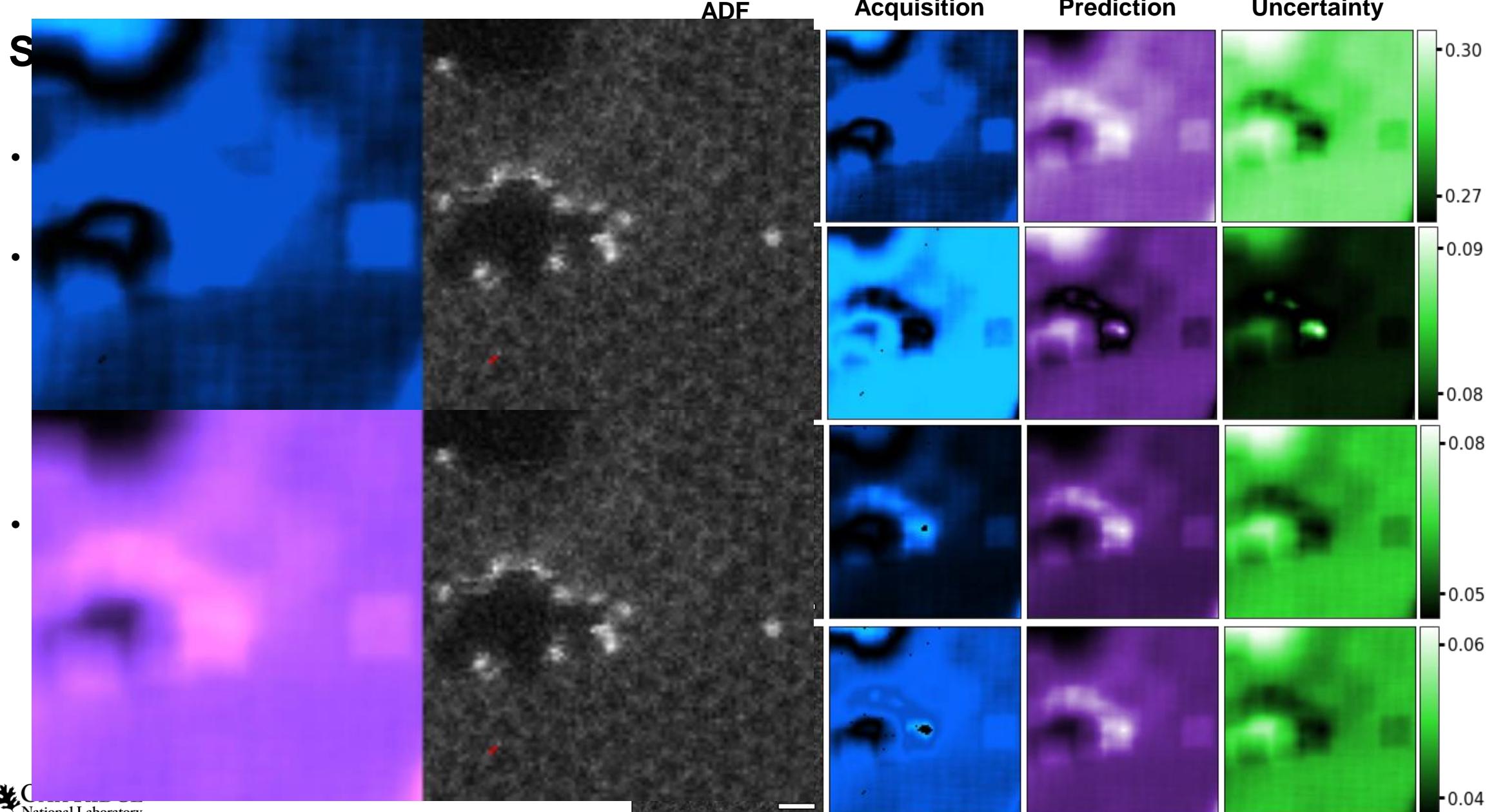


DKL: 4D STEM active learning (pre-acquired data)

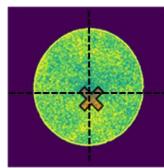
Scalarizer: |CoM|



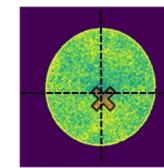
Active learning: real automated experiment



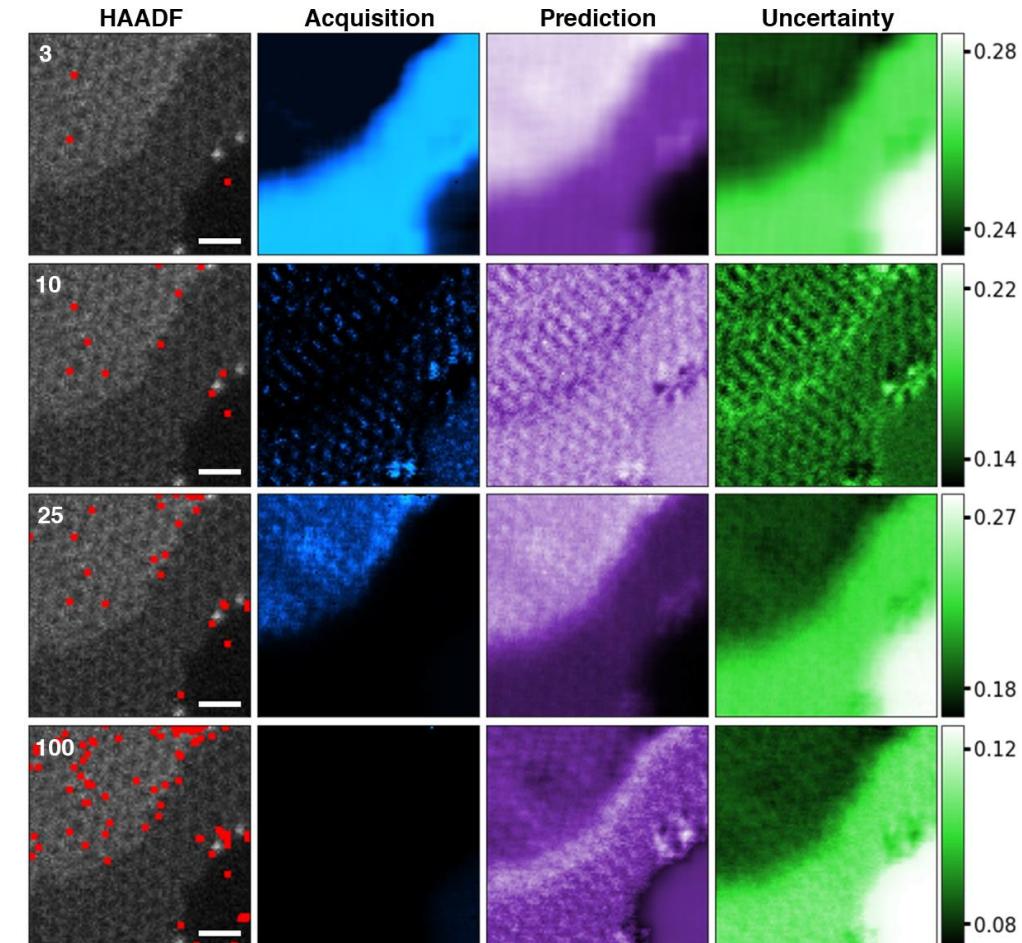
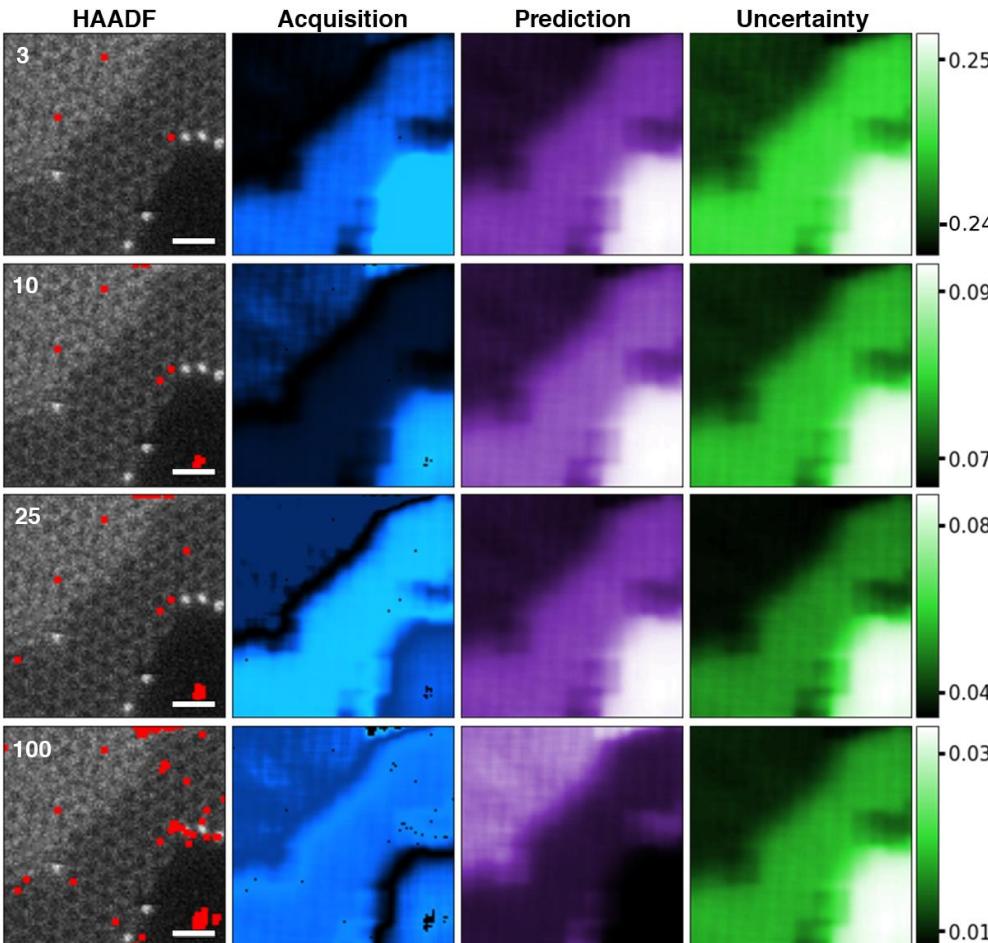
Effect of physics choice (scalarizer) on acquisition



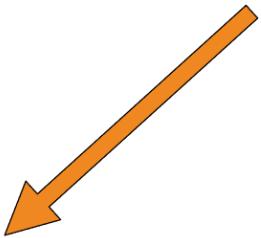
CoM
magnitude



CoM
angle

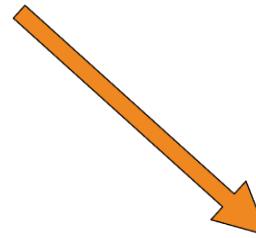


4D STEM



Center of mass

- Electric field
- Potential
- charge density

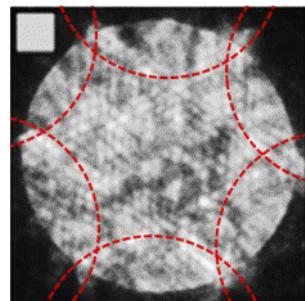
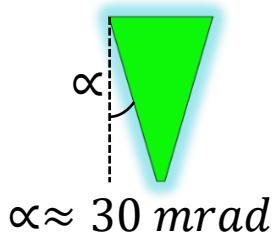


NBED disc fitting

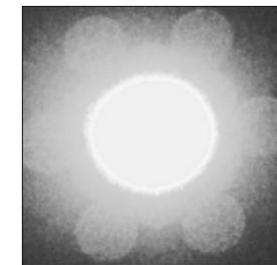
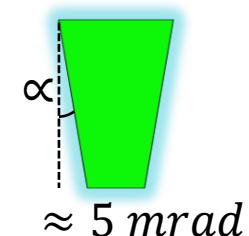
- Strain
- Grains

Require different probe conditions

Overlapping diffraction discs

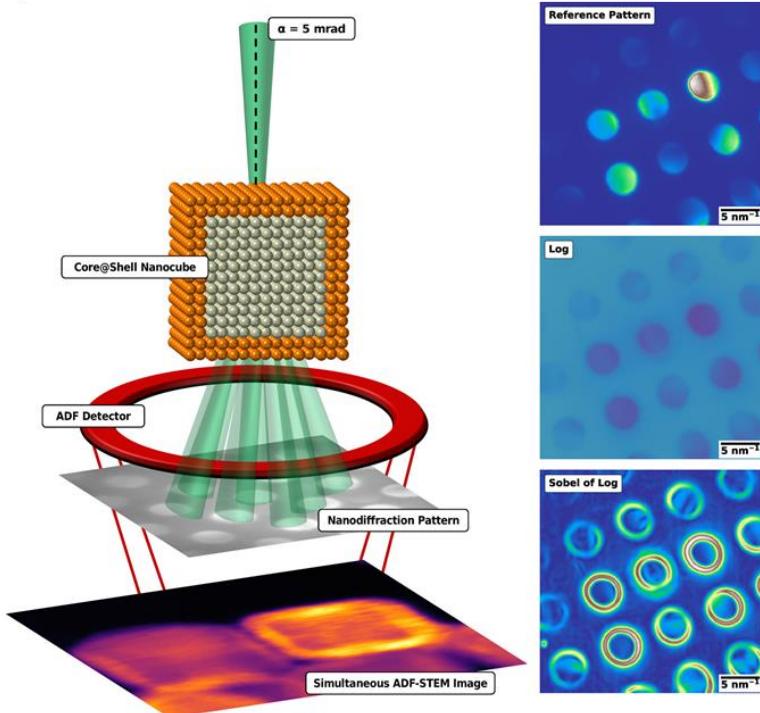


Non-overlapping diffraction discs



Measuring strain in the STEM

Using local diffraction

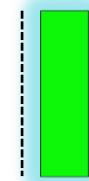


Mukherjee D. et. al., ACS Catal. 2020, 10, 10, 5529–5541

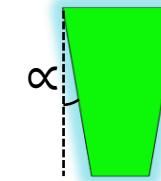
Requirements:

- Pixelated detector
- Diffraction **disc** fitting

Strain is calculated **after** experiment (offline analysis)

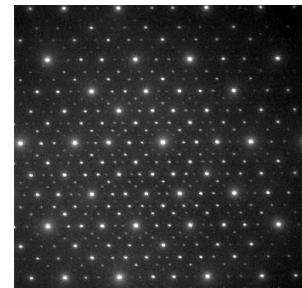


Parallel illumination



$\approx 5 \text{ mrad}$

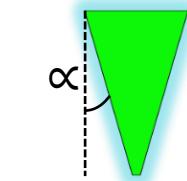
Convergent beam



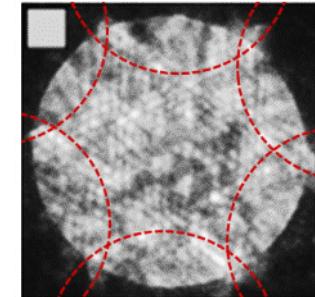
(Conventional TEM conditions)



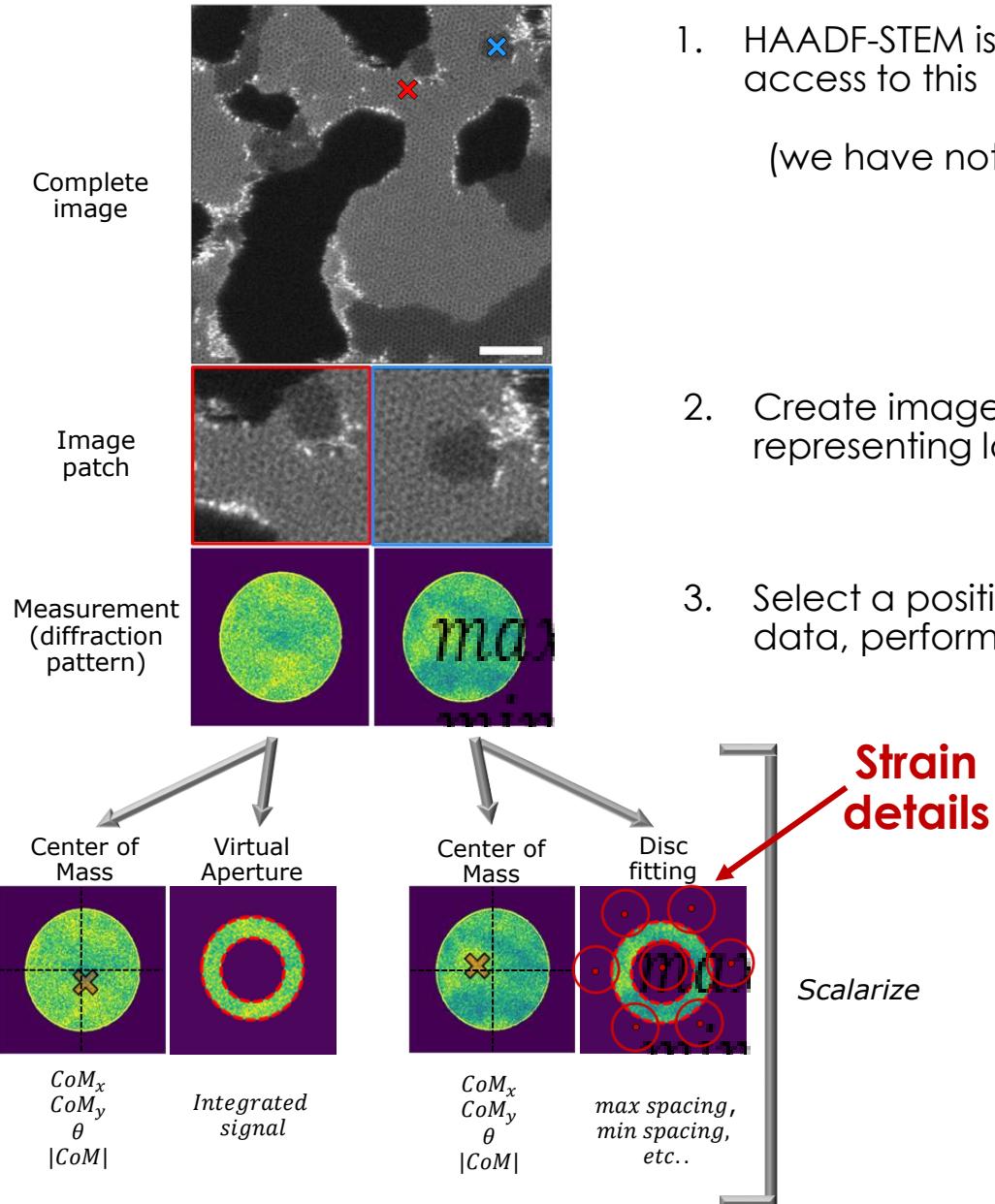
Strain, differential phase contrast (DPC), electric field, atomic potential maps



$\approx 30 \text{ mrad}$



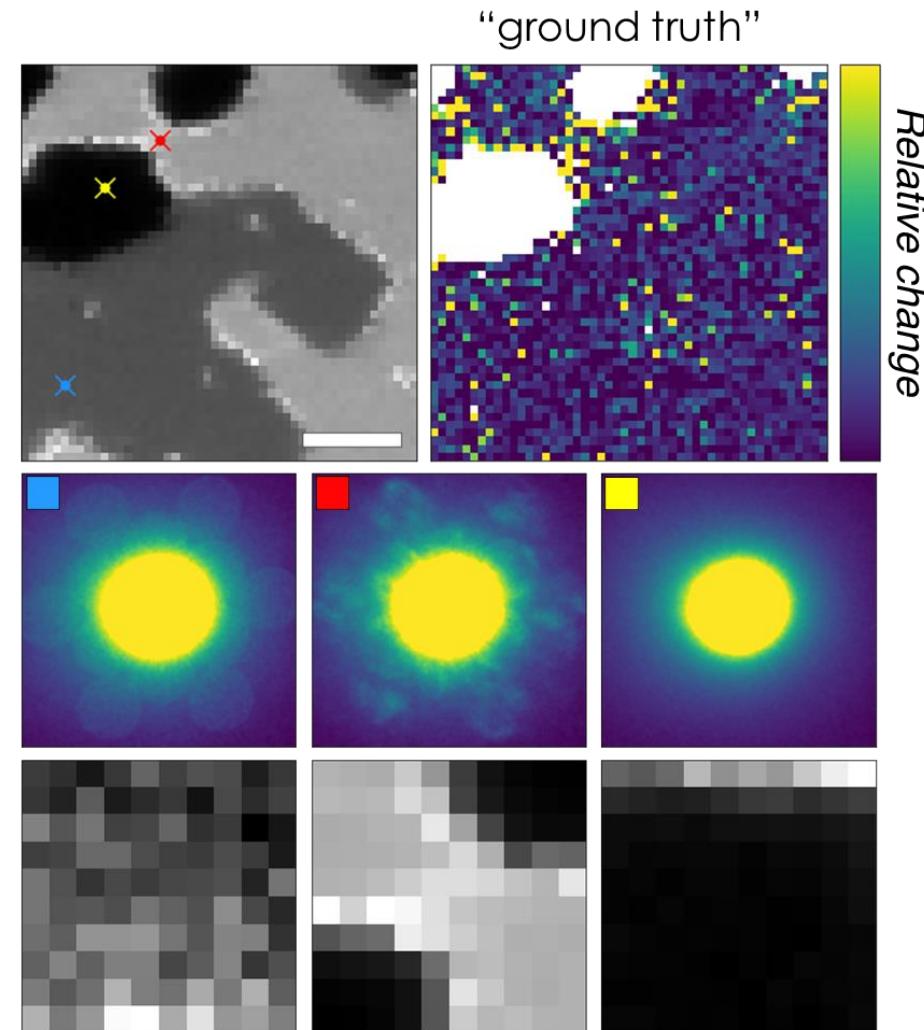
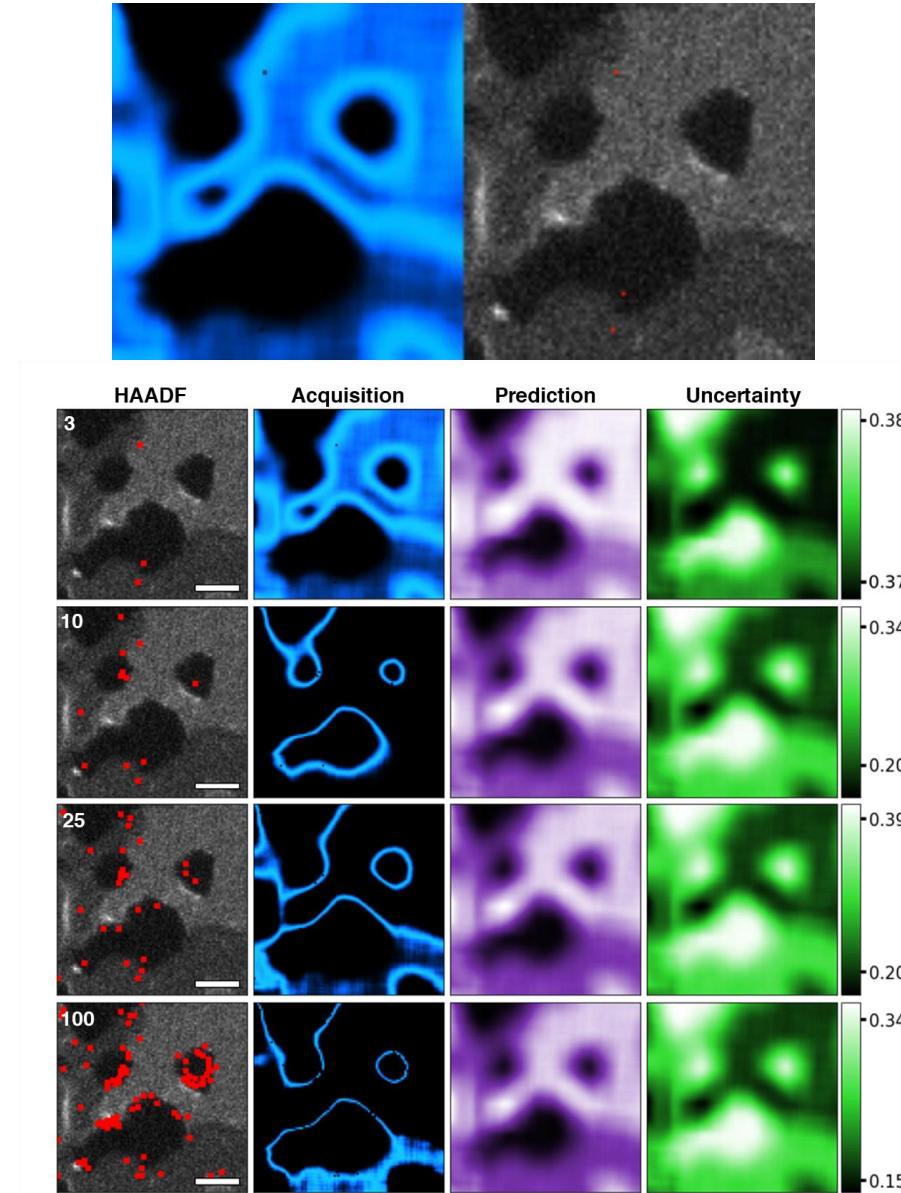
Neural Network Architecture



1. HAADF-STEM is acquired first, we have **complete** access to this
(we have not yet collected diffraction data yet)
2. Create image patches at each pixel, representing local geometry/structure
3. Select a position in space to acquire diffraction data, perform computation (scalarize).
4. Train / update network with **image patch** and **scalarized quantity PAIR**. Repeat 3&4 (automated)

This is controlled by physics-based “acquisition function”

Automated Experiments in 4D STEM: strain



Colab!

- Change to **GPU runtime** if not already
- from skimage.draw import **disk** (not circle)
- spectralavg = **128**
- dklgp = aoi.models.dklGPR(data_dim, embedim=2, precision="**=single**")