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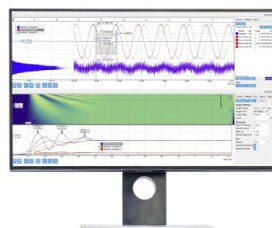
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# A multicenter numerical integration scheme for polyatomic molecules

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We propose a simple scheme for decomposition of molecular functions into single-center components. The problem of three-dimensional integration in molecular systems thus reduces to a sum of one-center, atomic-like integrations which are treated using standard numerical techniques in spherical polar coordinates. The resulting method is tested on representative diatomic and polyatomic systems for which we obtain five- or six-figure accuracy using a few thousand integration points per atom.

## I. INTRODUCTION

For many decades now, quantum chemists have relied on finite basis-set expansions as their primary computational tool. LCAO (linear combination of atomic orbitals) is currently the only viable general approach to *ab initio* computational chemistry. This dependence on finite basis sets has, however, undesirable consequences. A major part of virtually every quantum chemical publication is devoted to analysis of the basis sets employed in the calculations, reflecting the nagging and ever-present specter of basis-set incompleteness. Moreover, the inevitable discussion of basis-set details is, at least from a fundamental point of view, rather uninteresting.

Attempts have been made in recent years to escape from the conventional LCAO framework. In the special case of diatomic molecules, completely numerical, non-basis-set calculations are now possible on discrete two-dimensional meshes in prolate spheroidal coordinates. The first calculations of this numerical type were carried out in 1981 by the present author (see Ref. 1) on first-row diatomic systems in the Hartree-Fock-Slater or  $X\alpha$  approximation.<sup>2</sup> Since then, our method has evolved significantly and has recently been applied to dimers as heavy as  $\text{Cr}_2$  using various contemporary density functional theories.<sup>3,4</sup> A similar method has also been developed by Pyykkö and co-workers<sup>5</sup> with applications to a great variety of quantum chemical formalisms such as Hartree-Fock,  $X\alpha$ , MCSCF, and Dirac-Slater.<sup>6</sup> Completely numerical diatomic calculations are admittedly costly, but by now essentially routine. Note, however, that numerical computations are perfectly suited for vector and parallel processing. As parallel computing technology advances, therefore, discrete numerical techniques will become increasingly attractive.

It is our desire to develop a completely numerical, non-basis-set scheme for quantum chemical calculations on polyatomic molecules, in general. No such scheme currently exists (at least in *real* space, but see Ref. 7 for a proposed method in *momentum* space). In the present work, we take an initial step in this direction. Considered here is the fundamental problem of discrete numerical integration in multicenter systems.

Even if the ultimate objective of completely basis-set-free quantum chemistry is not soon realized, the ability to carry out numerical integrations in polyatomic molecules is

itself of immediate value. This is due to the increasing importance in quantum chemistry of so-called *density functional* theories, in which exchange and/or correlation energies of many-electron systems are approximated by three-dimensional integrals of local exchange-correlation densities derived from electron gas theory. The power of density functional methods has been aptly demonstrated in recent applications to diatomic, organometallic, and transition metal binuclear complexes in Refs. 4, 8, and 9, respectively, and the interested reader will find general reviews of the theoretical framework in Ref. 10. The reader should appreciate that the exchange-correlation integrals of density functional theories cannot be evaluated analytically, even in an LCAO context, and therein lies the importance of numerical integration algorithms.

In Sec. II, then, we outline the basic philosophy of the present integration scheme, and also discuss its extension to problems beyond the integration problem. Then, in Secs. III and IV, the method is described in detail. In Sec. V, the scheme is tested on the diatomic systems  $\text{H}_2$ ,  $\text{N}_2$ , and  $\text{P}_2$ , and on seven representative polyatomic systems of various geometries using integrands of density functional type. Diatomic systems are included for testing purposes because "exact" results for arbitrary integrands are obtainable from our existing numerical diatomic program.<sup>1</sup> Finally, we offer concluding remarks in Sec. VI. We include, also, an Appendix in which the reader will find certain details of secondary importance.

## II. GENERAL PHILOSOPHY

We are concerned with three-dimensional molecular integrals of the type

$$I = \int F(\mathbf{r}) d^3\mathbf{r}, \quad (1)$$

where  $F(\mathbf{r})$  is an arbitrary integrand. These may be approximated by discrete numerical summations of the form

$$I = \sum_i A_i F(\mathbf{r}_i), \quad (2)$$

where the  $\mathbf{r}_i$  and  $A_i$  are discrete integration points and their respective integration weights whose optimum distribution is the subject of the present work.

In molecules, of course, the integrand  $F(\mathbf{r})$  is dominated by cusps at atomic nuclei. Obviously, then, a straightforward

discrete integration in Cartesian coordinates is not recommended, though a marginally successful such scheme has nevertheless been proposed.<sup>11</sup> Alternatively, standard orthogonal *curvilinear* coordinates exist for the special case of two-center systems<sup>12</sup> and these have been fully exploited in the previously-cited work of Becke<sup>1</sup> and Pyykkö *et al.*<sup>5</sup> on diatomic molecules. For systems containing three or more centers, however, standard multicenter coordinates do not exist, and we therefore face a difficult problem.

An obvious and popular solution to this multicenter integration problem involves partitioning of molecular space into discrete regions or "cells" of simple geometry commensurate with the positions of the nuclei, such that straightforward numerical integration may be carried out within each region. This cellular approach has been developed to a high degree of sophistication in the recent work of Boerrigter, te Velde, and Baerends.<sup>13</sup> Cellular methods require, however, considerable intuition and input regarding the disposition of the various regions. A scheme requiring knowledge only of the positions (and perhaps charges) of the atomic nuclei, such as proposed in the present communication, would be somewhat more convenient.

In the context of cellular methods, the work of Bader and co-workers (see, e.g., Ref. 14 and references therein) is also relevant. Bader has defined atomic fragments in molecular systems on the basis of the topology of the total electron density. These atomic fragments have fascinating theoretical significance and ingenious numerical techniques have been devised for integrating within them.<sup>15</sup> Though the primary applications of Bader's formalism are theoretical and interpretative, some of the associated numerical techniques could conceivably be adopted for general-purpose molecular integration as well.

We should also acknowledge the method of Ellis and Painter<sup>16</sup> which has played such an important role in the development of the Hartree-Fock-Slater molecular structure program of Baerends and co-workers (see Ref. 17 and references therein). The method of Ellis and Painter, based on Diophantine theory,<sup>18</sup> behaves rather poorly as the number of integration points increases, however, and is therefore unsuitable for integration of high accuracy. The reader is referred, e.g., to a comparison with the method of Boerrigter *et al.* in Ref. 13.

The approach of the present work is based on a very simple philosophy. We assume that a *relative weight* function  $w_n(\mathbf{r})$  can be assigned to each nucleus  $n$  in the system such that, for all  $\mathbf{r}$ ,

$$\sum_n w_n(\mathbf{r}) = 1 \quad (3)$$

and such that each  $w_n(\mathbf{r})$  has value *unity* in the vicinity of its *own* nucleus, but *vanishes* in a continuous and well-behaved manner near *any other* nucleus. In a sense, then, the system is divided not into conventional discrete cells, but into *fuzzy, overlapping, analytically continuous* cells instead. This image of a "fuzzy" cellular partitioning is not intended merely as a visual aid, but will, in fact, be implemented quite literally in Sec. III.

An arbitrary molecular function  $F(\mathbf{r})$  may thus be de-

composed into single-center components  $F_n(\mathbf{r})$  as follows:

$$F(\mathbf{r}) = \sum_n F_n(\mathbf{r}), \quad (4)$$

where

$$F_n(\mathbf{r}) = w_n(\mathbf{r})F(\mathbf{r}) \quad (5)$$

and the multicenter integration of Eq. (1) therefore reduces to a sum of *single-center* integrations  $I_n$  over each of the nuclei in the system:

$$I = \sum_n I_n, \quad (6)$$

where

$$I_n = \int F_n(\mathbf{r}) d^3\mathbf{r}. \quad (7)$$

If the weight functions  $w_n(\mathbf{r})$  are sufficiently well behaved, as defined more precisely in Sec. III, then each atomic subintegration  $I_n$  may be carried out using standard single-center numerical techniques in spherical polar coordinates.

This philosophy is not, in a general sense, particularly new. A conceptually similar approach has also been described by Boys and Rajagopal.<sup>19</sup> The present work, however, and particularly the present decomposition scheme, differs considerably in detail.

Also, we wish to point out that the present decomposition scheme extends naturally to the problem of solving Poisson's equation:

$$\nabla^2 V_{\text{el}}(\mathbf{r}) = -4\pi\rho(\mathbf{r}) \quad (8)$$

for the Coulomb potential  $V_{\text{el}}(\mathbf{r})$  due to an electron density distribution  $\rho(\mathbf{r})$ . Fully numerical solution of Poisson's equation, though tractable in diatomic systems as a result of the work of Becke<sup>1</sup> and Pyykkö *et al.*,<sup>5</sup> is not yet feasible generally. If, however, the proposed atomic decomposition scheme is applied to the density distribution  $\rho(\mathbf{r})$ , then Eq. (8), being linear, reduces to a set of independent *one-center* equations, each of which may then be solved using standard single-center techniques such as spherical harmonic analysis, etc. Fully numerical solution of Poisson's equation, in combination with a good numerical integration scheme (such as the scheme proposed here), would allow basis-set-free computation of molecular Coulomb interaction energies and matrix elements. We are currently pursuing this problem and hope to discuss it further in later publications.

### III. NUCLEAR WEIGHT FUNCTIONS

We now describe a novel procedure for generating the nuclear weight functions  $w_n(\mathbf{r})$  introduced in Sec. II. We begin by considering a conventional cellular partitioning of molecular space into so-called Voronoi polyhedra. Then, we "soften" the boundaries of the resulting polyhedra to create the "fuzzy cells" to which we alluded above.

Given an arbitrary distribution of atomic centers in three-dimensional space, a unique polyhedron may be associated with each nucleus by a relatively simple prescription. Consider, e.g., nucleus  $i$  as reference, and construct for every other nucleus  $j \neq i$  the perpendicular bisecting plane of the vector joining  $i$  to  $j$ . Clearly, the minimum volume enclosed

by all these bisecting planes is a *polyhedron* containing all points nearer to nucleus  $i$  than to any other. These unique polyhedra, one for each  $i$ , perfectly fill the space of the molecule and are called Voronoi polyhedra (or Voronoi cells). In the special case of a *periodic* atomic lattice, these are somewhat better known as Wigner-Seitz cells.

Efficient algorithms for the automatic construction of Voronoi polyhedra have been published (see, e.g., Ref. 20 and references therein), but we shall introduce a rather unconventional scheme which is uniquely appropriate for the present purposes.

Our scheme involves the popular two-center coordinate system known as confocal elliptical coordinates  $(\lambda, \mu, \phi)$ . The coordinate  $\phi$  denotes angle about the internuclear axis, and coordinates  $\lambda$  and  $\mu$  are defined by

$$\lambda = \frac{r_1 + r_2}{R_{12}}, \quad \mu = \frac{r_1 - r_2}{R_{12}}, \quad (9)$$

where  $r_1, r_2$ , and  $R_{12}$  denote distance to nucleus 1, distance to nucleus 2, and the internuclear separation, respectively. The coordinate  $\lambda$  labels ellipsoids of revolution with foci at the positions of the nuclei, and  $\mu$  similarly labels hyperboloids. Their ranges are indicated below:

$$\begin{aligned} 0 &\leq \phi \leq 2\pi, \\ 1 &\leq \lambda < \infty, \\ -1 &\leq \mu \leq +1. \end{aligned} \quad (10)$$

Notice that the surface  $\mu = 0$  corresponds to the perpendicular bisector of the internuclear axis, just as required in the construction of Voronoi polyhedra. Also, we note that  $\mu = -1$  corresponds to a semi-infinite ray extending from center 1 along the nuclear axis to infinity, and that  $\mu = +1$  corresponds to a similar ray from center 2 to infinity.

Elliptical coordinates furnish a natural and convenient definition of Voronoi polyhedra in an interesting analytical sense. With center  $i$  as reference, consider in turn each of the other centers  $j \neq i$  and establish elliptical coordinates  $\lambda_{ij}, \mu_{ij}$ , and  $\phi_{ij}$  on the foci  $i$  and  $j$ . Of special interest is the coordinate

$$\mu_{ij} = \frac{r_i - r_j}{R_{ij}}, \quad (11)$$

where  $r_i$  and  $r_j$  denote distances to nuclei  $i$  and  $j$ , respectively, and  $R_{ij}$  is the internuclear separation. With this hyperboloidal coordinate as argument, consider the step function

$$s(\mu_{ij}) = \begin{cases} 1, & -1 \leq \mu_{ij} \leq 0 \\ 0, & 0 < \mu_{ij} \leq +1 \end{cases} \quad (12)$$

and then, recalling that the surface  $\mu_{ij} = 0$  is the perpendicular bisector of  $R_{ij}$ , observe that the Voronoi polyhedron on nucleus  $i$  is defined by the following simple product:

$$P_i(\mathbf{r}) = \prod_{j \neq i} s(\mu_{ij}). \quad (13)$$

This product, which will hereafter be called a "cell function," has value unity if  $\mathbf{r}$  lies inside the cell, and vanishes if  $\mathbf{r}$  lies outside.

The above definition of Voronoi cells suggests an obvious generalization. Let us replace the step function of Eq. (12) with an appropriate *continuous* analog, resembling, e.g., the finite-temperature Fermi function of statistical me-

chanics. In other words, let us soften or "smooth out" the step discontinuity at  $\mu_{ij} = 0$ . The cell functions of Eq. (13) will now, as a consequence, represent *analytically continuous, mutually overlapping* regions, and we thereby create the fuzzy Voronoi polyhedra visualized earlier.

Having thus expanded its original definition to include continuous analogs, we will hereafter refer to  $s(\mu_{ij})$  as a "cutoff function" or a "cutoff profile." Possible cutoff profiles are not, of course, unique, but we shall impose the following minimum requirements:

$$\begin{aligned} s(-1) &= 1, \\ s(+1) &= 0, \end{aligned} \quad (14)$$

$$\frac{ds}{d\mu}(-1) = \frac{ds}{d\mu}(+1) = 0. \quad (15)$$

The derivative constraints of Eq. (15) ensure that  $s(\mu_{ij})$  does not have nuclear cusps. After considerable experimentation with a variety of possible functional forms, including rational, exponential, and trigonometric forms, we have found a particularly useful representation involving simple polynomials only. We describe this representation below.

Consider, first of all, the related problem of finding an odd function  $f(\mu)$  in the interval  $-1 \leq \mu \leq +1$  with the properties

$$\begin{aligned} f(-1) &= -1, \\ f(+1) &= +1, \end{aligned} \quad (16)$$

$$\frac{df}{d\mu}(-1) = \frac{df}{d\mu}(+1) = 0, \quad (17)$$

from which the cutoff function  $s(\mu)$  is obtained by the transformation

$$s(\mu) = \frac{1}{2}[1 - f(\mu)]. \quad (18)$$

The simplest possible  $f(\mu)$  satisfying the above constraints is a two-term polynomial:

$$p(\mu) = \frac{3}{2}\mu - \frac{1}{2}\mu^3. \quad (19)$$

This simple polynomial varies smoothly between the end-points  $-1$  and  $+1$ , but, for reasons discussed momentarily, is not "sharp" enough (i.e., not "step function-like" enough) for our purposes. If, however, we *iterate* as follows:

$$\begin{aligned} f_1(\mu) &= p(\mu), \\ f_2(\mu) &= p[p(\mu)], \\ f_3(\mu) &= p[p[p(\mu)]], \dots, \end{aligned} \quad (20)$$

then successively sharper functions may be generated with increasing iteration order  $k$ . For extremely large  $k$ , in fact, a step function is ultimately evolved.

A useful sequence of cutoff profiles  $s_k(\mu)$  is thus generated from the above sequence of  $f_k(\mu)$  by Eq. (18):

$$s_k(\mu) = \frac{1}{2}[1 - f_k(\mu)]. \quad (21)$$

We face, however, the problem of determining the optimum value of  $k$ . It is here that the stipulation "sufficiently well behaved," invoked somewhat vaguely in Sec. II, must be satisfied. If  $k$  is too small, then  $s_k(\mu_{ij})$  will not sufficiently extinguish the nuclear cusp on center  $j$ . If  $k$  is too large, on

the other hand, the step function of Eq. (12) is recovered and we revert, therefore, to conventional discrete cells. Our sequence of cutoff profiles is illustrated in Fig. 1 for values of  $k$  from 1 to 5. We have found, on the basis of our experience thus far, that the value  $k = 3$  is appropriate for general applications. This particular choice has therefore been used in all calculations of the present work, and is convincingly supported by the quality of our results in Sec. V.

The above polynomial cutoff profiles strictly satisfy the constraints of Eq. (14). Therefore, the  $i$ th cell function  $P_i(\mathbf{r})$ , in accordance with the definition of Eq. (13), has value *unity* at nucleus  $i$ , but *vanishes* at all other nuclei  $j \neq i$  and also on each semi-infinite ray extending from  $j$  to infinity along the  $ij$  axis. These cell functions are obvious candidates for the nuclear weights  $w_n(\mathbf{r})$ . In order to satisfy Eq. (3), however, we use the following definition:

$$w_n(\mathbf{r}) = P_n(\mathbf{r}) / \sum_m P_m(\mathbf{r}), \quad (22)$$

where the summation over  $m$  in the denominator includes all nuclei in the system (i.e.,  $m = n$  as well).

Our description of the atomic decomposition scheme is now essentially complete, insofar, at least, as *homonuclear* systems (i.e., atoms of one kind only) are concerned. In *heteronuclear* systems, though, the scheme would obviously benefit from an ability to accommodate varying atomic sizes. Atomic size adjustments are indeed possible, and have, in fact, been incorporated. We wish to proceed at this time, however, to a description of the single-center atomic subintegrations, and therefore, we defer discussion of atomic size adjustments to the Appendix.

#### IV. SINGLE-CENTER SUBINTEGRATIONS

Having discussed the atomic decomposition scheme, we now describe our numerical procedure for computing the single-center integrals  $I_n$  of Eq. (7). The following description applies independently to each nucleus in the molecule.

If we establish a spherical polar coordinate system  $(r, \vartheta, \phi)$  on nucleus  $n$ , then the volume integral of the single-center component  $F_n(\mathbf{r})$  is given by the familiar expression

$$I_n = \iiint F_n(r, \vartheta, \phi) r^2 \sin \vartheta \, dr \, d\vartheta \, d\phi. \quad (23)$$

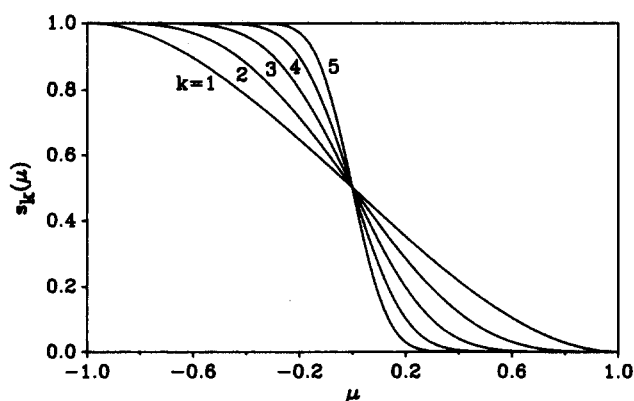


FIG. 1. Cutoff profiles  $s_k(\mu)$  of Eq. (21) for  $k = 1$  to 5.

This three-dimensional integration may be carried out by treating each of the  $r$ ,  $\vartheta$ , and  $\phi$  coordinates independently. There exists, however, a small but impressive literature on Gauss-type quadrature formulas for direct two-dimensional integration on the surface of the unit sphere (see Refs. 21 to 28) and we therefore write

$$I_n = \iint F_n(r, \Omega) r^2 \, dr \, d\Omega, \quad (24)$$

where  $\Omega$  denotes solid angle. Integration on the surface of the unit sphere and integration over solid angle are, of course, equivalent. In subsequent paragraphs, then, we consider first of all the indicated solid-angle integration and then we discuss the radial part.

Numerical integration on the surface of the unit sphere is a problem of fundamental interest. Following the important early work of McLaren,<sup>21</sup> Stroud,<sup>22</sup> and Sobolev,<sup>23</sup> quadrature formulas of remarkably high order (up to  $l = 35$ , where all spherical harmonics of order  $l$  or less are integrated exactly) have been published by Lebedev<sup>24-27</sup> and by Konyaev.<sup>28</sup> We have coded Lebedev's quadratures of orders 11, 17, and 23 containing 50, 110, and 194 integration points, respectively (Refs. 24 and 25) for the investigations of the present work. The resulting point distributions are invariant with respect to the octahedral group. The point distributions of Konyaev,<sup>28</sup> on the other hand, have icosahedral symmetry.

The reader may wonder if such large angular meshes are really necessary, or, indeed, if they need be even larger. Excessively large meshes as implied by the integration of off-center nuclear cusps are clearly not required. Nevertheless, our cutoff profiles  $s(\mu_{ij})$  display significant step-function-like features in internuclear regions, and, as a consequence, moderately large angular meshes are indeed necessary. Our experience thus far indicates that something in the order of 100 or 200 angular points is appropriate for five- or six-figure integration accuracy (see Sec. V).

The radial integrations are performed using standard Gauss-Chebyshev quadrature of the second kind.<sup>29</sup> We prefer Gauss-Chebyshev to the much more popular Gauss-Legendre quadrature because Chebyshev points and weights are given by simple analytical formulas. First, however, the standard Gauss-Chebyshev integration interval  $-1 \leq x \leq +1$  must be mapped into the semi-infinite radial interval  $0 \leq r < \infty$ . This is accomplished by the following coordinate transformation:

$$r = r_m \frac{(1+x)}{(1-x)}, \quad (25)$$

where  $r_m$  is a parameter corresponding to the midpoint of the integration interval at  $x = 0$ . This parameter allows adjustment of the radial point distribution to a suitable physical scale. In the present work,  $r_m$  is chosen as *half* of the *Bragg-Slater radius* of the respective atom (see Ref. 30 and also the Appendix), except for hydrogen in which case the factor 1/2 is not applied.

The number of radial points on each center is difficult to assign uniquely and depends, of course, on desired precision and on the particular atom in question. Nevertheless, the following empirical rule of thumb appears to provide a rela-

tive integration accuracy in the order of  $10^{-5}$ . The hydrogen atom receives an initial quota of 20 radial points, and, thereafter, an additional 5 points are assigned for each additional atomic shell. This rule of thumb assigns, e.g., 25 radial points to the first-row atoms Li to Ne, and 30 points to the second-row atoms Na to Ar, etc. Greater or fewer points are appropriate, of course, if relative integration accuracy greater or less than  $10^{-5}$  is required.

The above single-center procedure is repeated for every nucleus in the system. The overall scheme is extremely simple in principle and in practice. In Sec. V it is tested on representative diatomic and polyatomic molecules.

## V. DIATOMIC AND POLYATOMIC TESTS

The tests reported here are not exhaustive. We simply wish to demonstrate the viability of the present method and to assess the sort of accuracy obtainable. Our first series of tests is performed on *diatomic* systems in order that various integrals of the density functional type may be compared with exact results from our existing numerical diatomic program.<sup>1</sup>

We consider specifically the diatomic systems  $H_2$ ,  $N_2$ , and  $P_2$  at their equilibrium internuclear separations. Furthermore, we choose as test integrands the functionals

$$F(\mathbf{r}) = \rho, \rho^{4/3}, \rho^{5/3}, V_{el}\rho, \text{ and } V_{nuc}\rho, \quad (26)$$

where  $\rho$  is the total electron density,  $V_{el}$  is the corresponding Coulomb potential, and  $V_{nuc}$  is the potential due to the atomic nuclei. The second and third of the above integrands,  $\rho^{4/3}$  and  $\rho^{5/3}$ , are well-known density functional approximations for exchange and kinetic energy densities, respectively, while the final integrands  $V_{el}\rho$  and  $V_{nuc}\rho$  are components of the potential energy. In the above selection of integrands, therefore, all major components of the total energy of a many-electron system are represented.

For convenience, however, we model the total density  $\rho$  simply as a sum of free atomic densities obtained from the Hartree-Fock orbitals of Clementi and Roetti.<sup>31</sup> This approximation simplifies the present tests tremendously, but is not expected to bias our ultimate assessment in any way. Within this model, the value of the potential  $V_{el}$  at each of the integration mesh points is calculated analytically.

In diatomic systems, of course, axial symmetry reduces our problem from three dimensions to only two. Full solid-angle integration as described in Sec. IV is therefore unnecessary. We wish, nevertheless, to estimate the potential accuracy of the present integration scheme in general polyatomic situations, and therefore we ignore axial symmetry in these diatomic tests. In Tables I to III, numerical integrals of the five functionals of Eq. (26) are presented for 50-point and for 110-point angular meshes. The associated radial meshes are the same in both cases, with the number of radial points given by the rules of Sec. IV. We observe that an overall accuracy in the order of five or six significant figures is obtained, and that all five integrands, despite their qualitative differences, are equally well treated.

Further results are presented in Table IV for the numerical integration of the total electronic charge in seven polyatomic systems representing a wide variety of typical

TABLE I. Numerical integration in  $H_2$ .

Integrand	Exact	Angular mesh points	
		50	110
$\rho$	2.000 00	1.999 98	2.000 00
$\rho^{4/3}$	0.660 758	0.660 746	0.660 751
$\rho^{5/3}$	0.257 193	0.257 190	0.257 190
$V_{el}\rho$	2.257 04	2.257 00	2.257 02
$ V_{nuc}\rho $	3.220 08	3.220 05	3.220 07

molecular geometries. Again we assume, for convenience only, that the total molecular charge density is just a sum of atomic Hartree-Fock densities. However, the disposition of the single-center subintegration meshes in polyatomic systems is slightly more complicated than for the previous dimers. This is due to the fact that somewhat finer meshes are required for *central* atoms in polyatomic molecules than for ligand atoms or atoms in linear coordination. We therefore assign roughly twice as many angular points to the central atoms in the molecules of Table IV (except for C in linear  $CO_2$ ) than are assigned to the ligand atoms. Also, each central atom is assigned ten additional *radial* points as well. These *ad hoc* rules for assigning mesh points are nothing more than rough guides in determining efficient point distributions. They are not rigorously based, nor should they be viewed as corrupting the fundamental simplicity of the present integration scheme.

Results are given in Table IV for two classes of integration meshes, coarse meshes containing 50-point angular components (110 for central atoms) and somewhat finer meshes containing 110-point angular parts (194 for central atoms). The same *radial* meshes are used in either case, in accordance with the rules of Sec. IV and the *ad hoc* addition of ten radial points for central atoms. The resulting charge integrals display a relative accuracy in the order of  $10^{-4}$  in the case of the coarse meshes, and  $10^{-5}$  in the case of the finer meshes. In view of the previous diatomic results, we infer that similar precision would be obtained as well for the other integrands in the list of Eq. (26).

The recent cellular method of Boerrigter *et al.*<sup>13</sup> is of high accuracy also, but the great simplicity of the present scheme and its relatively straightforward extension to solution of Poisson's equation (see Sec. II) are strong advantages. Still higher accuracy may, of course, be achieved by increasing the number of quadrature points even further.

TABLE II. Numerical integration in  $N_2$ .

Integrand	Exact	Angular mesh points	
		50	110
$\rho$	14.0001	14.0003	14.0000
$\rho^{4/3}$	15.9961	15.9963	15.9961
$\rho^{5/3}$	34.3728	34.3733	34.3729
$V_{el}\rho$	148.988	148.990	148.988
$ V_{nuc}\rho $	303.106	303.115	303.109



TABLE III. Numerical integration in  $P_2$ .

Integrand	Exact	Angular mesh points	
		50	110
$\rho$	30.0001	29.9993	30.0001
$\rho^{4/3}$	56.3134	56.3143	56.3135
$\rho^{5/3}$	220.002	220.004	220.001
$V_{el}\rho$	737.814	737.818	737.814
$ V_{nuc}\rho $	1749.64	1749.74	1749.73

Note, however, that extremely high precision is not, in fact, required in calculating most quantities of chemical interest. Chemically interesting quantities, such as dissociation energies, can normally be expressed as *differences* between molecular and atomic data. If, therefore, molecular and atomic reference calculations are performed consistently, then cancellation of numerical errors obviates the need for results of high absolute precision. Cancellation of numerical errors has been very useful in our earlier diatomic work<sup>1</sup> and will no doubt assist our future efforts as well.

## VI. CONCLUSIONS

In the present communication, we have addressed the problem of numerical integration in molecular systems by decomposing the integrand into single-center terms, each of which may then be integrated by standard single-center numerical techniques. Our basic decomposition scheme may also be applied to solution of Poisson's equation (work in progress) and eventually, we hope, to the computation of numerical molecular orbitals. This work is only the first step in the development of a comprehensive program system for nonbasis-set quantum chemical calculations. Alternatively, the present scheme may be usefully incorporated into existing density functional LCAO programs such as those of Baerends *et al.*,<sup>17</sup> Dunlap *et al.*,<sup>32</sup> etc., all of which rely to some degree on numerical integration techniques.

The tests of Sec. V demonstrate that the present scheme is both accurate and practical. Results of high precision (five or six figures) are obtained with a few thousand integration points per atom. Notice, also, that the necessary number of points may be reduced dramatically in molecules of appro-

priate geometries if the respective octahedral and icosahedral symmetries of Lebedev's<sup>24-27</sup> and Konyaev's<sup>28</sup> angular point distributions are exploited.

The present results are very encouraging. Further investigations and applications of this novel polyatomic integration scheme will be reported in future communications.

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## APPENDIX: ATOMIC SIZE ADJUSTMENTS

To simplify the following discussion, the terminology cell "face" or cell "boundary" will be used somewhat loosely. Our fuzzy cells do not, of course, have discrete faces. We shall, however, use these terms to denote the surfaces on which the cutoff functions  $s(\mu_{ij})$  take the intermediate value 1/2.

The faces of the Voronoi polyhedra described in Sec. III bisect the internuclear axes exactly. In *heteronuclear* systems, however, variations in atomic size do not necessarily favor such a "democratic" partitioning. We desire, therefore, to shift our cell boundaries away from the internuclear midpoints in heteronuclear molecules.

First of all, we adopt as a definition of atomic size the empirical atomic radii of Bragg and Slater.<sup>30</sup> Bragg-Slater radii are derived from the observation that interatomic bond lengths in solids and molecules, whether ionic or covalent, are approximately equal to pairwise sums of unique atomic radii. This observation may be rationalized by the further observation that these unique atomic radii correlate remarkably well with the positions of valence-shell probability-density maxima obtained independently from theoretical calculations on free atoms. In the present work, Bragg-Slater radii as tabulated in Ref. 30 are therefore used as guides in the adjustment of cell boundaries (and also the adjustment of the radial integration parameters  $r_m$  of Sec. IV). For hydrogen, however, we prefer a radius of 0.35 Å, somewhat larger than Slater's value of 0.25 Å. Details of the boundary adjustment procedure are now outlined.

A cell boundary is defined by the surface on which the cutoff function  $s(\mu_{ij})$  has value 1/2, or, equivalently, the surface on which its argument  $\mu_{ij}$  has value 0. As emphasized in Sec. III, the surfaces  $\mu_{ij} = 0$  are perpendicular bisectors of internuclear axes, and thus the faces of the cells discussed so far lie exactly at internuclear midpoints. Our cell boundaries can be shifted *off center*, however, by a simple transformation of coordinates from  $\mu_{ij}$  to a new coordinate  $\nu_{ij}$ . In other words, a heteronuclear cutoff profile  $s_{het}(\mu_{ij})$  can be defined in the following manner:

$$s_{het}(\mu_{ij}) = s(\nu_{ij}), \quad (A1)$$

where the new argument  $\nu_{ij}$  is related, in the present work, to the old argument  $\mu_{ij}$  by

$$\nu_{ij} = \mu_{ij} + a_{ij}(1 - \mu_{ij}^2) \quad (A2)$$

TABLE IV. Charge integration in polyatomic systems.

Molecule	Exact	Angular mesh points	
		50 <sup>a</sup>	110 <sup>b</sup>
CO <sub>2</sub>	22.0001	22.0004	21.9995
H <sub>2</sub> O	10.0000	10.0006	10.0001
BF <sub>3</sub>	31.9999	32.0003	32.0003
NH <sub>3</sub>	10.0000	10.0013	10.0001
CH <sub>4</sub>	10.0000	10.0020	10.0002
PF <sub>5</sub>	59.9998	60.0029	59.9986
SF <sub>6</sub>	69.9997	70.0051	69.9995

<sup>a</sup> 110 for central atoms, plus 10 additional radial points.

<sup>b</sup> 194 for central atoms, plus 10 additional radial points.

and where  $a_{ij}$  is a parameter with value in the range

$$|a_{ij}| \leq \frac{1}{2} \quad (\text{A3})$$

to ensure monotonicity. We thus transform our cell boundaries from bisecting planes to hyperboloidal surfaces  $v_{ij} = 0$ , towards center  $i$  or towards center  $j$  depending on the sign of  $a_{ij}$ . Though other transformations are also conceivable, Eq. (A2) is particularly simple.

The above transformation is easily incorporated into the cell-construction algorithm of Sec. III. We simply require for each nuclear pair  $i$  and  $j$ , with respective Bragg-Slater radii  $R_i$  and  $R_j$  and internuclear separation  $R_{ij}$ , a value for parameter  $a_{ij}$ . In order to specify  $a_{ij}$ , consider the point at which the internuclear  $ij$  axis intersects the *shifted* boundary surface  $v_{ij} = 0$ , and let us *assume* that the ratio of  $r_i$  to  $r_j$  (the distances to the respective nuclei) is equal to the *ratio of the Bragg-Slater radii*:

$$\frac{r_i}{r_j} = \frac{R_i}{R_j} \equiv \chi. \quad (\text{A4})$$

Parameter  $a_{ij}$  is then determined by the condition  $v_{ij} = 0$  as follows:

$$a_{ij} = \frac{u_{ij}}{u_{ij}^2 - 1}, \quad (\text{A5})$$

where  $u_{ij}$  is related to the ratio  $\chi$  of the Bragg-Slater radii by

$$u_{ij} = \frac{r_i - r_j}{R_{ij}} = \frac{r_i - r_j}{r_i + r_j} = \frac{\chi - 1}{\chi + 1}. \quad (\text{A6})$$

Should, however, the resulting  $a_{ij}$  fall outside the range allowed by Eq. (A3), then we simply assign the appropriate endpoint value  $\pm 1/2$ . This occurs if the Bragg-Slater radii  $R_i$  and  $R_j$  differ by a factor greater than roughly 2.4. The transformation of Eq. (A2) does not, therefore, provide unlimited freedom in shifting cell boundaries, but the available flexibility is apparently quite adequate.

The adjusted partitioning scheme described in this section gives significantly better results than obtained from "homonuclear" decompositions. All of the polyatomic integrals discussed in Sec. V have therefore been calculated using shifted cell boundaries.

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