# Analysis of Niobium Electropolishing Using a Generalized Distribution of Relaxation Times Method

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Using electrochemical impedance spectroscopy, we have devised a method of sensing the microscopic surface conditions on the surface of niobium as it is undergoing an electrochemical polishing (EP) treatment. The method uses electrochemical impedance spectroscopy (EIS) to gather information on the surface state of the electrode without disrupting the polishing reaction. The EIS data is analyzed using a so-called distribution of relaxation times (DRT) method. Using DRT, the EIS data can be deconvolved into discrete relaxation time peaks without any apriori knowledge of the electrode dynamics. By analyzing the relaxation time peaks, we are able to distinguish two distinct modes of the EP reaction. As the polishing voltage is increased, the electrode transitions from the low voltage EP mode, characterized by a single relaxation time peaks, to the high voltage EP mode, characterized by two relaxation time peaks. We theorize that this second peak is caused by the formation of an oxide layer on the electrode. We also find that this oxide induced peak transitions from to a negative relaxation time, which is indicative of a blocking electrode process. By analyzing EPed samples, we show that samples polished in the low voltage mode have significantly higher surface roughness due to grain etching and faceting. We find that the surface roughness of the samples only improves when the oxide film peak is present and in the negative relaxation time region. This shows that EIS combined with DRT analysis can be used to predict etching on EPed Nb. This method can also be performed before or during the EP, which could allow for adjustment of polishing parameters to guarantee a smooth cavity surface finish.

# I. INTRODUCTION

Electropolishing (EP) is commonly used to polish Nb SRF cavities to nanometer scale surface roughness.

EIS has been used to study the chemistry of niobium in HF electrolytes before.[1–4] However, these studies have only analyzed the spectrum using qualitative methods or traditional equivalent circuit fitting techniques. These methods have proven insufficient for explaining several important phenomenon of EP such as surface etching and spontaneous current oscillations.

In this study, we use a model-free method of analyzing the EIS spectrum called distribution of relaxation times (DRT) analysis. The advantage of this method is that the EIS spectrum can be easily characterized over a large range of polishing conditions without making any assumptions about the underlying chemical processes.[5–7] This method has been used to characterize a wide range of electrochemical systems such as fuel cells,[8–10] batteries,[11–13] and supercapacitors.[14]

Using this method, we are able to observe the formation of the  $Nb_2O_5$  layer as the polishing voltage is increased. We also show that the formation of the oxide corresponds with a reduction in surface etching by analyzing Nb samples using confocal laser microscopy.

### II. EXPERIMENTAL

The electrolyte used for this experiment is composed of a 9:1 ratio of 98% by volume sulfuric acid and 70% by volume hydroflouric acid. A strip of Nb foil was prepared by cleaning with alcohol. Approximately 8.4 cm<sup>2</sup> surface

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area of the strip was exposed to the electrolyte. A Nb wire was used as a pseudo-electrode, since it is resistant to the EP electrolyte. The counter electrode is an aluminum rod. The temperature of the electrolyte was held at 21 °C using an aluminum cooling coil immersed in the electrolyte. The EIS measurements were performed using a BioLogic VSP-300 potentiostat over a frequency range of  $0.5\,\mathrm{Hz}$  to  $2\times10^5\,\mathrm{Hz}$ . The EIS measurement was repeated for polishing voltages ranging from  $0.5\,\mathrm{V}$  to  $0.9\,\mathrm{V}$  measured versus the open circuit potential.

A series of Nb samples were electropolished using voltages ranging from  $0.5\,\mathrm{V}$  to  $1.2\,\mathrm{V}$  to determine the effect of the polishing voltage on the surface finish of the Nb.  $5\,\mu\mathrm{m}$  of material was removed from each sample. To measure the surface finish we use a Keyence VX-K 3000 confocal laser microscope. The confocal microscope produces a 2D height map of the samples.

### III. RESULTS

EIS measurements reveal a complex evolution of the surface chemistry as the polishing voltage is increased through the  $0.5~\rm V$  to  $1.0~\rm V$  range. As shown in figure 1, at  $0.5~\rm V$  the system exhibits a single capacitive process. This is most likely caused by the polarization of the electrical double-layer that occurs between the Nb electrode and the electrolyte. At this low voltage, the Nb<sub>2</sub>O<sub>5</sub> has not been formed due to the low oxidation current and high concentration of HF at the surface of the electrode.

When the voltage is increased, two new features appear, an inductive loop at mid-frequencies and a second capacitive loop at low frequencies. This low frequency loop as caused by the formation of the oxide film and depletion of the HF near the electrode surface. This limits the current through the electrode at higher voltages.

When the voltage is increased from 0.78 V to 0.86 V the low frequency loop changes from decreasing impedance with increasing frequency to increasing as shown in figure 1.

The capacitive loop with increasing real impedance is indicative of a surface blocking process. This type of process is typically seen in electroplating processes containing a blocking species. This phenomenon is associated with brightening of the surface, since the blocking species preferentially blocks deposition on surface peaks.

#### IV. OXIDE FILM FORMATION DURING NIOBIUM ELECTROPOLISHING

The dissolution of niobium is caused by its reaction with HF and its ions in the electrolyte. However, before the niobium can be dissolved it must be oxidized. When the rate of oxidation exceeds the maximum rate of dissolution by HF, limited by diffusion of HF to the electrode surface, niobium oxides can form on the surface. The idea of a compact oxide film present during electropolishing is not a new one.[2, 3] Theories such as the point defect model have been used to explain the behavior of niobium and other pasivating metals[15–19] Ranjith, et. al.[4] propose a reaction model for niobium involving two intermediate oxides, Nb<sup>2+</sup> and Nb<sup>5+</sup> and using this model qualitatively predict the impedance behavior in both the active and passive regions including the negative differential resistance which we have reported in our own measurements. A diagram of this reaction model is shown if figure 3. It is this model that we will use to explain our EIS measurements of niobium electropolishing.

One important aspect that the simple reaction model does not account for is the inhomogeneity of the niobium electrode. When oxides begin to form on the surface, they do not cover the entire surface. Due to slight differences in the oxidation reaction activation energy across the surface, caused by differing grain orientations, surface defects, and the local curvature, the oxide forms preferentially in certain areas. The surface has a mixed oxide with a certain fraction covered by  $Nb_2O_5$ , NbO, and Nb. As the surface potential is increased a larger portion of the electrode is covered by  $Nb_2O_5$ . Once the entire surface is covered, the dissolution rate of the surface becomes homogeneous and polishing can occur. If the polishing voltage is not sufficiently high across the surface of the niobium, some areas will experience etching instead of polishing. This effect can be seen in 2. On samples polished at low voltage, some of the grains have a facetted surface indicating that the niobium is directly exposed to the electrolyte in these regions leading to grain orientation specific effects. This faceting disappears when the polishing voltage is above  $0.8 \, \text{V}$ . This effect coincides with the appearance of the negative resistive locus in the nyquist diagram 1, C.

### V. GENERALIZED DISTRIBUTION OF RELAXATION TIMES

To better quantify the formation of the continuous oxide film, we employ an analysis method known as the distribution of relaxation times (DRT). The DRT is a flexible and general method that can be used to model complex impedance spectrum and extract useful information without the use of a traditional equivalent circuit model. Using

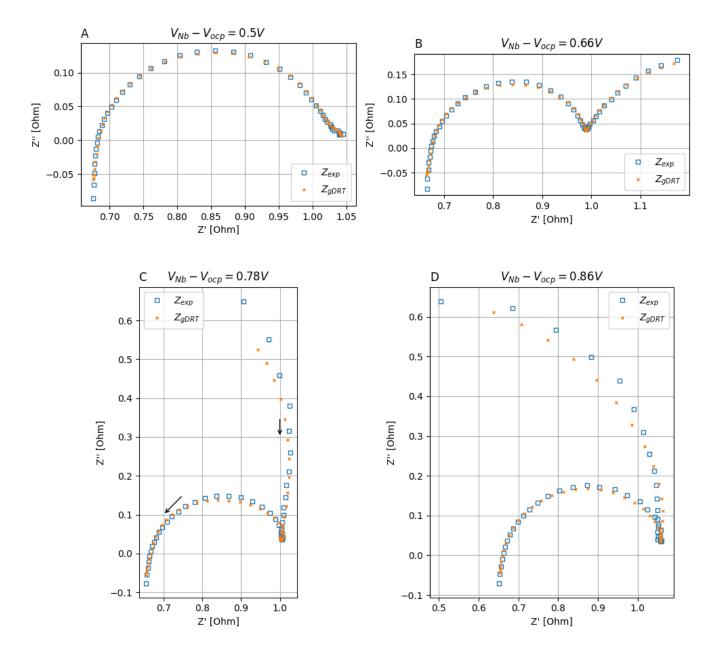


FIG. 1. The complex impedance of Nb. Each plot shows the different regimes of electropolishing at low voltage. A one capacitive loop, B two capacitive loops, C capacitive, inductive, and capacitive loops, D capacitive, inductive, and capacitive loop with a negative resistance. The arrows indicate the direction of increasing frequency.

this method we can more accurately identify the occurrence of new electrode reactions such as the creation of the NbO and  $Nb_2O_5$  oxides. We can therefore identify at what polishing voltages these oxidation reactions occur.

Traditional distribution of relaxation time analysis is based on an assumption that the electrode reaction can be modeled by a series of voigt circuits, a resistor and a capacitor connected in parallel. By adding an infinite number of infinitesimal voigt elements, any arbitrary capacitive impedance can be described. The impedance of these voigt elements is described by the distribution of relaxation times function.

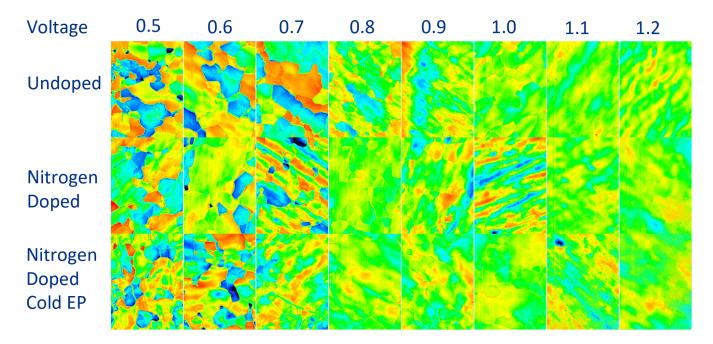


FIG. 2. The surface height maps of Nb samples electropolished at different voltages and temperatures.

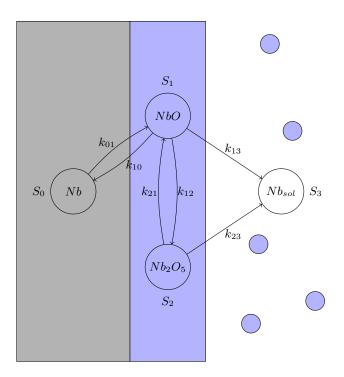


FIG. 3. Proposed mechanism for niobium electropolishing in HF containing electrolyte by a two step oxidation reaction.

$$Z_{RC}\left(\omega\right) = \frac{R}{1 + j\omega RC}\tag{1}$$

$$Z_{voigt}(\omega) = R_s + \sum_{n=1}^{N} \frac{R_n}{1 + j\frac{\omega}{\omega_n}}$$
 (2)

$$Z_{DRT}(\omega) = R_s + \int_0^\infty \frac{G(\omega_0)d\omega_0}{1 + j\frac{\omega}{\omega_0}}$$
(3)

This model is sufficient to describe any capacitive-like processes, i.e. those with positive real impedance and negative imaginary impedance. However, this does not include the impedance response of niobium electropolishing due to the negative time constant locus at low frequency as seen in figure 1 D. We also observe an inductive mid-frequency loop as seen in figure 1 C and D. To model this feature, we must generalize the DRT function to include all four quadrants of the complex impedance.

This generalization requires the unintuitive addition of negative capacitances, alternatively inductances, and negative resistances to the DRT transmission line model. However, these circuit elements do not violate any physical laws, since the circuit model is just a representation of the electrochemical system and no physical significance should be interpreted from the circuit itself. We can show this to be true by deriving the circuit model from the kinetic equations of the electrochemical system. The electrochemical system can be described by a state vector whose evolution over time is described by a system of first order, potentially non-linear, differential equations. Wu, et. al show that for a small perturbation, the impedance spectrum of such a system can always be represented as a sum of voigt-like elements determined by the kinetic equations of the system and both positive and negative capacitance and resistance are allowed. [20, 21]

$$Z_F(\omega) = R_t + \sum_{n=1}^{N} \frac{k_n}{j\omega - s_n}$$
(4)

$$Z_{gDRT}(\omega) = R_t + \int_{-\infty}^{\infty} \frac{k(s) ds}{j\omega - s}$$
 (5)

$$k\left(s\right) = \omega_0 G\left(\omega_0\right) \tag{6}$$

$$s = -\omega_0 \tag{7}$$

$$Z_{gDRT}(\omega) = R_t + \int_{-\infty}^{\infty} \frac{G(\omega_0) d\omega_0}{1 + j\frac{\omega}{\omega_0}}$$
(8)

If we let  $s_i = \frac{-1}{R_i C_i}$  and  $k_i = \frac{1}{C_i}$  we see that this equation describes a classical voigt element. However, in this equation the constants  $s_i$  and  $k_i$  can take on both positive and negative values. We can construct a transmission line model with an infinite number of these voigt-like elements in the same way as for the derivation of the classical DRT function, the only difference being the distribution spans both positive and negative frequencies and the function  $G(\omega_0)$  is free to take both positive and negative values. By generalizing the DRT method in this manner, it can in principle fit any impedance spectrum.

## VI. DISTRIBUTION OF RELAXATION TIMES OF NIOBIUM ELECTROPOLISHING

Using the generalized DRT method, we are able to fit the Nb impedance spectrum across all frequencies including the inductive mid-frequency loop and the low frequency loop as seen in figure 4. From figure 5 we can see that there are three prominent peaks in the distribution of relaxations times. The peak centered around 100 Hz to 1000 Hz corresponds to the electric double layer capacitance. The center of this peak shifts to the right linearly with the polishing voltage. This shows that the relaxation time of the electric double layer follows an Arrhenius type relation.

The two other peaks centered around 10 Hz, one with a positive time constant and one with a negative time constant, most likely correspond to niobium oxidation reactions. The intensity of the oxidation peaks increases at higher polishing voltages indicating that a larger fraction of the niobium is covered by oxides. At a polishing voltage of 0.75 V the third peak appears with a negative time constant. A negative time constant indicates that this oxidation reaction has a blocking effect on the reaction. This blocking effect leads to a current plateau as well as a negative differential resistance above 0.8 V.

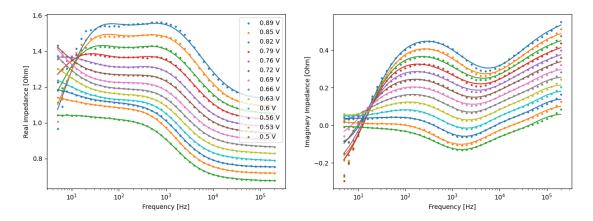


FIG. 4. The complex impedance of Nb in electropolishing electrolyte measured using EIS. Each curve is offset from zero.

#### VII. DISCUSSION

Comparing figure 5 to the electropolished samples shown in figure 2 we can see that the appearance of the third peak coincides with a significant change in the surface of the electropolished niobium. The surface goes from an etched surface with facets and large grain boundary steps to a much more polished finish. We theorize that this transition is caused by the stabilization of the oxide film which occurs when the HF is depleted on the niobium electrode surface and the surface oxide transitions from a thin non-homogeneous layer to a thick homogeneous one. Since the oxide film is homogeneous, the effects of grain orientation on the dissolution reaction are eliminated. The result is a smoothing effect on the niobium. At polishing voltages below this critical voltage, the oxide film is too thin or does not cover the entire niobium surface, which leads to etching.

It is much more difficult to control the polishing voltage when electropolishing cavities compared to small scale samples. This is due to the much larger current and more complex geometry of the cavity and cathode. Currently, cavity electropolishing machines do not employ a reference electrode and instead only set a fixed voltage between the anode and cathode. This means that cathode losses and ohmic losses in the electrolyte can significantly alter the polishing potential and even create a non-homogeneous potential distribution over the surface of the cavity. This is why etching can occur even when applying much higher potentials typically used during cavity EP.

To combat the uncertainty of electropolishing cavities it may be possible to apply EIS measurements in the production environment. This would allow for a live diagnostic of the polishing conditions inside the cavity without interfering with the regular EP workflow. This information can be used to tweak the EP parameters such as voltage and temperature to ensure that the cavity is in an optimal polishing regime. Using this technique it may be possible to electropolish more complex cavity geometries such as quarter-wave and half-wave geometries.

# VIII. CONCLUSION

In this study we have shown that the etching phenomenon seen in electropolished SRF cavities is caused by insufficient polishing voltage. Niobium polished below a critical voltage will experience etching whereas niobium polished above this voltage will experience smoothing. We theorize that the cause of this change in surface finish is a transition from a partially formed niobium oxide film to a fully formed, stable oxide film. This theory is evidenced by measurements of the niobium electropolishing impedance spectrum over a range of polishing voltages. The impedance spectrum indicates the presence of a blocking mechanism that limits the current. This blocking mechanism is most likely caused by the oxide film which forms when HF is depleted near the niobium surface and the oxidation rate exceeds the rate of dissolution by HF.

### IX. SUPPLEMENTAL INFORMATION

This Supplemental section will cover the mathematical process of calculating the generalized DRT from experimental measurements of the impedance spectrum. To find the DRT function numerically, we must first discretize the function

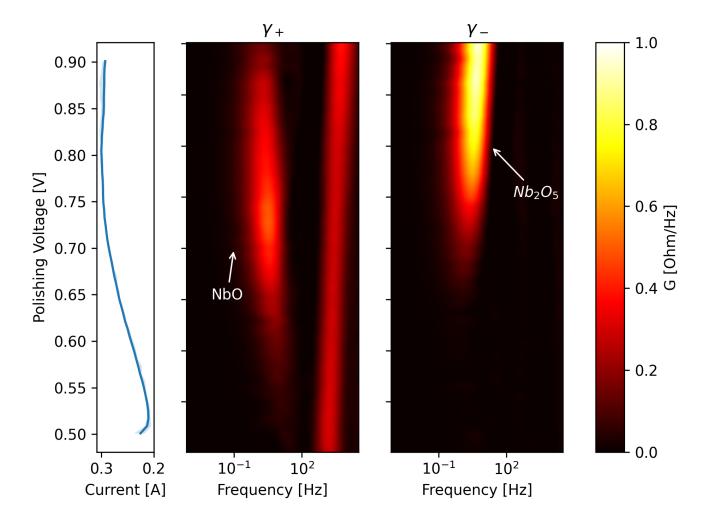


FIG. 5. The Distribution of relaxation times calculated from the EIS impedance data for each of the measured potentials. The current-voltage curve is shown in the graph on the left.

using a finite set of basis functions which will approximate the continuous DRT function. This is done by first transforming the problem into the much more natural log space. For a chosen basis function, the integral can then be calculated numerically and the problem is then transformed into a simple case of linear regression. We must also apply a normalization procedure to our regression to prevent overfitting.

# A. Discretizing the Integral of the Generalized DRT Function in Log Coordinates

It is necessary to discretize the DRT function in the log based coordinate system since EIS measurements are typically performed over many orders of magnitude in frequency. Otherwise, the computational accuracy will be too low at the low frequencies, or too high at the high frequencies. However, this procedure is more challenging for the generalized DRT model, since the integral over  $\omega_0$  spans both positive and negative values. To solve this problem we must separate the integral in equation 5 into the sum of two integrals each integrating over one side of the real axis. After a change of variables and reversing the direction of integration we can substitute  $G(\omega_0)$  with a piece wise fit function  $\gamma_{\pm}$  ( $\ln \omega_0$ ).

$$Z_{gDRT} = R_t + \int_{-\infty}^{0} \frac{G(\omega_0)d\omega_0}{1 + j\frac{\omega}{\omega_0}} + \int_{0}^{\infty} \frac{G(\omega_0)d\omega_0}{1 + j\frac{\omega}{\omega_0}}$$

$$\tag{9}$$

$$Z_{gDRT} = R_t + \int_0^\infty \frac{G(\omega_0)d\omega_0}{1 + j\frac{\omega}{\omega_0}} + \frac{G(-\omega_0)d\omega_0}{1 - j\frac{\omega}{\omega_0}}$$
(10)

$$G(\omega_0) d\omega_0 = \begin{cases} \gamma_+ (\ln \omega_0) d \ln \omega_0 & \omega_0 \ge 0\\ \gamma_- (\ln -\omega_0) d \ln \omega_0 & \omega_0 \le 0 \end{cases}$$
(11)

$$Z_{gDRT} = R_t + \int_{-\infty}^{\infty} \frac{\gamma_+ (\ln \omega_0) d \ln \omega_0}{1 + j \frac{\omega}{\omega_0}} + \frac{\gamma_- (\ln \omega_0) d \ln \omega_0}{1 - j \frac{\omega}{\omega_0}}$$
(12)

To solve for the functions  $\gamma_{\pm}(\ln \omega_0)$  numerically we approximate the function using a basis function  $\phi_n(\ln \omega_0)$ .

$$\gamma_{\pm}(\ln \omega_0) \approx \sum_{n=1}^{N} x_n^{\pm} \phi_n(\ln \omega_0)$$
(13)

$$Z_{gDRT} = R_t + \int_{-\infty}^{\infty} \sum_{n=1}^{N} \frac{x_n^+ \phi_n(\ln \omega_0) d\ln \omega_0}{1 + j\frac{\omega}{\omega_0}} + \sum_{n=1}^{N} \frac{x_n^- \phi_n(\ln \omega_0) d\ln \omega_0}{1 - j\frac{\omega}{\omega_0}}$$
(14)

Using algebraic manipulation and a change of variables, the integral can be separated into a real part and an imaginary part.

$$x_n^{re} = x_n^+ + x_n^-$$
 (15)

$$x_n^{im} = x_n^+ - x_n^- \tag{16}$$

$$Z_{gDRT} = R_t + \sum_{n=1}^{N} x_n^{re} \int_{-\infty}^{\infty} \frac{\phi_n(\ln \omega_0) d \ln \omega_0}{1 + \left(\frac{\omega}{\omega_0}\right)^2} + j \sum_{n=1}^{N} x_n^{im} \int_{-\infty}^{\infty} \frac{\frac{\omega}{\omega_0} \phi_n(\ln \omega_0) d \ln \omega_0}{1 + \left(\frac{\omega}{\omega_0}\right)^2}$$
(17)

The impedance at a frequency  $\omega = \omega_m$  is a linear function of the values of  $x_n^{re}$  and  $x_n^{im}$ .

$$Z_{gDRT}(\omega_m) = R_t + \sum_{n=1}^{N} x_n^{re} A_{n,m}^{re} + \sum_{n=1}^{N} x_n^{im} A_{n,m}^{im}$$
(18)

$$A_{n,m}^{re} = \int_{-\infty}^{\infty} \frac{\phi_n(\ln \omega_0) d \ln \omega_0}{1 + \left(\frac{\omega_m}{\omega_0}\right)^2}$$
(19)

$$A_{n,m}^{im} = \int_{-\infty}^{\infty} \frac{\frac{\omega_m}{\omega_0} \phi_n(\ln \omega_0) d \ln \omega_0}{1 + \left(\frac{\omega_m}{\omega_0}\right)^2}$$
 (20)

Given a set of M experimentally measured impedance data points,  $\mathbf{Z_{exp}} = [Z_1, \dots, Z_M]$  at frequencies  $\omega = [\omega_1, \dots, \omega_M]$  we can find the values of  $x_n^{re}$  and  $x_n^{im}$  using linear regression.

$$\min_{\mathbf{x^{re}, x^{im}}} \|\mathbf{Z_{gDRT}} - \mathbf{Z_{exp}}\| = \min_{\mathbf{x^{re}, x^{im}}} (\|\mathbf{x^{re}A^{re}} - Re(\mathbf{Z_{exp}})\| + \|\mathbf{x^{im}A^{im}} - Im(\mathbf{Z_{exp}})\|)$$
(21)

# B. Normalization

When finding the DRT function from experimental data, a normalization method is required to prevent overfitting. In this study we use a modified Tikhonov regularization, based on the work of Wan, et. al. [6], that penalizes the derivatives of the DRT function. This has the effect of smoothing the function and eliminating oscillations caused by

overfitting. With the regularization term added the objective becomes to minimize the residuals and the normalization term together.

$$\min_{\mathbf{x}^{re}, \mathbf{x}^{im}} \left( \| \mathbf{x}^{re} \mathbf{A}^{re} - Re \left( \mathbf{Z}_{exp} \right) \| + \| \mathbf{x}^{im} \mathbf{A}^{im} - Im \left( \mathbf{Z}_{exp} \right) \| + \mathbf{x}^{re} \mathbf{M} \mathbf{x}^{reT} + \mathbf{x}^{im} \mathbf{M} \mathbf{x}^{imT} \right)$$
(22)

The matrix **M** is calculated by integrating the square of the derivatives of the test functions,  $\phi_n(\omega_0)$ . Calculating the zeroth derivative, i.e. the function itself, is equivalent to standard Tikhonov regularization. The strength of the regularization is controlled by multiplying by a constant  $\lambda_k$ , where k is the k-th derivative of the test functions.

$$(\mathbf{M}_k)_{n,m} = \int_{-\infty}^{\infty} \frac{d^k \phi_n}{dln\tau^k} \frac{d^k \phi_m}{dln\tau^k} dln\tau$$
 (23)

$$\mathbf{M} = \sum_{k=0}^{K} \lambda_k \mathbf{M}_k \tag{24}$$

The optimum values of  $\lambda_k$  are difficult to find mathematically, so values were manually adjusted up to k=2 to maximize the fitting strength without sacrificing the accuracy of the fit. Mathematical heuristics such as the L-curve, cross-validation, Fourier transform[22] or Bayesian methods[23].

### C. Basis Function

The choice of basis function is quite arbitrary as long as the resulting function space is large enough to contain the exact solution for the DRT function. However, in practice the basis function has a large impact on the number of parameters required and the level of regularization required. The basis function should also be relatively easy to compute using a computer to speed up the computation and differentiable so that the regularization matrix can be calculated. A natural choice is to use the log Gaussian, which has been shown to be effective at fitting the DRT function for several kinds of impedance systems[6].

$$\phi_n(\ln \omega) = \frac{1}{\sigma\sqrt{2\pi}} e^{-\left(\frac{\ln \omega_0 - \ln \omega_n}{\sigma\sqrt{2}}\right)^2} \tag{25}$$

Plugging this test function into equation 19 and 20 gives the following expressions for the matrices  $A_{re}$  and  $A_{im}$ .

$$A_{n,m}^{re} = \frac{1}{\sigma\sqrt{2\pi}} \int_{-\infty}^{\infty} \frac{e^{-\left(\frac{\ln\omega_0 - \ln\omega_n}{\sigma\sqrt{2}}\right)^2} d\ln\omega_0}{1 + e^{2(\ln\omega_m - \ln\omega_0)}}$$
(26)

$$A_{n,m}^{im} = \frac{1}{\sigma\sqrt{2\pi}} \int_{-\infty}^{\infty} \frac{e^{\ln\omega_m - \ln\omega_0} e^{-\left(\frac{\ln\omega_0 - \ln\omega_n}{\sigma\sqrt{2}}\right)^2} d\ln\omega_0}{1 + e^{2(\ln\omega_m - \ln\omega_0)}}$$
(27)

<sup>[1]</sup> S. Cattarin, M. Musiani, and B. Tribollet, Nb electrodissolution in acid fluoride medium: steady-state and impedance investigations, Journal of The Electrochemical Society 149, B457 (2002).

<sup>[2]</sup> H. Tian, S. G. Corcoran, C. E. Reece, and M. J. Kelley, The mechanism of electropolishing of niobium in hydrofluoric–sulfuric acid electrolyte, Journal of The Electrochemical Society 155, D563 (2008).

<sup>[3]</sup> H. Tian, C. Reece, M. Kelley, and S. Corcoran, Novel characterization of the electropolishing of niobium with sulfuric and hydrofluoric acid mixtures, Tech. Rep. (Thomas Jefferson National Accelerator Facility (TJNAF), Newport News, VA..., 2008).

<sup>[4]</sup> P. M. Ranjith, M. T. Rao, S. Sapra, I. I. Suni, and R. Srinivasan, On the anodic dissolution of tantalum and niobium in hydrofluoric acid, Journal of The Electrochemical Society 165, C258 (2018).

<sup>[5]</sup> W. A. Yager, The Distribution of Relaxation Times in Typical Dielectrics, Physics 7, 434 (1936) https://pubs.aip.org/aip/jap/article-pdf/7/12/434/19667442/1\_1745355.pdf.

<sup>[6]</sup> T. H. Wan, M. Saccoccio, C. Chen, and F. Ciucci, Influence of the discretization methods on the distribution of relaxation times deconvolution: implementing radial basis functions with drttools, Electrochimica Acta 184, 483 (2015).

- [7] Y. Zhang, Y. Chen, M. Yan, and F. Chen, Reconstruction of relaxation time distribution from linear electrochemical impedance spectroscopy, Journal of Power Sources 283, 464 (2015).
- [8] V. Sonn, A. Leonide, and E. Ivers-Tiffée, Combined deconvolution and cnls fitting approach applied on the impedance response of technical ni8ysz cermet electrodes, Journal of The Electrochemical Society 155, B675 (2008).
- [9] H. Schichlein, A. C. Müller, M. Voigts, A. Krügel, and E. Ivers-Tiffée, Deconvolution of electrochemical impedance spectra for the identification of electrode reaction mechanisms in solid oxide fuel cells, Journal of Applied Electrochemistry 32, 875 (2002).
- [10] A. Leonide, V. Sonn, A. Weber, and E. Ivers-Tiffée, Evaluation and modeling of the cell resistance in anode-supported solid oxide fuel cells, Journal of The Electrochemical Society 155, B36 (2007).
- [11] J. P. Schmidt, P. Berg, M. Schönleber, A. Weber, and E. Ivers-Tiffée, The distribution of relaxation times as basis for generalized time-domain models for li-ion batteries, Journal of Power Sources 221, 70 (2013).
- [12] M. Hahn, S. Schindler, L.-C. Triebs, and M. A. Danzer, Optimized process parameters for a reproducible distribution of relaxation times analysis of electrochemical systems, Batteries 5, 10.3390/batteries5020043 (2019).
- [13] R. Soni, J. B. Robinson, P. R. Shearing, D. J. Brett, A. J. Rettie, and T. S. Miller, Lithium-sulfur battery diagnostics through distribution of relaxation times analysis, Energy Storage Materials 51, 97 (2022).
- [14] L. Helseth, Modelling supercapacitors using a dynamic equivalent circuit with a distribution of relaxation times, Journal of Energy Storage 25, 100912 (2019).
- [15] M. Bojinov, The ability of a surface charge approach to describe barrier film growth on tungsten in acidic solutions, Electrochimica Acta 42, 3489 (1997).
- [16] A. Girginov and M. Bojinov, Conduction mechanisms in the (+) valve metal/oxide film/electrolyte system, Journal of the University of Chemical Technology and Metallurgy 43, 29 (2008).
- [17] M. Bojinov, S. Cattarin, M. Musiani, and B. Tribollet, Evidence of coupling between film growth and metal dissolution in passivation processes, Electrochimica acta 48, 4107 (2003).
- [18] D. D. Macdonald and M. Urquidi-Macdonald, Theory of steady-state passive films, Journal of the Electrochemical Society 137, 2395 (1990).
- [19] D. D. Macdonald, S. R. Biaggio, and H. Song, Steady-state passive films: Interfacial kinetic effects and diagnostic criteria, Journal of the Electrochemical Society 139, 170 (1992).
- [20] X. Wu, H. Ma, and S. Chen, An investigation of faradaic admittance for electrode processes involving n state variables besides electrode potential, Journal of the Electrochemical Society 145, 517 (1998).
- [21] X. Wu, H. Ma, S. Chen, Z. Xu, and A. Sui, General equivalent circuits for faradaic electrode processes under electrochemical reaction control, Journal of the Electrochemical Society **146**, 1847 (1999).
- [22] B. A. Boukamp and A. Rolle, Analysis and application of distribution of relaxation times in solid state ionics, Solid State Ionics 302, 12 (2017), iSSFIT12.
- [23] F. Ciucci and C. Chen, Analysis of electrochemical impedance spectroscopy data using the distribution of relaxation times: A bayesian and hierarchical bayesian approach, Electrochimica Acta 167, 439 (2015).