

# **In introduction to molecular electron microscopy**

- Imaging macromolecular assemblies

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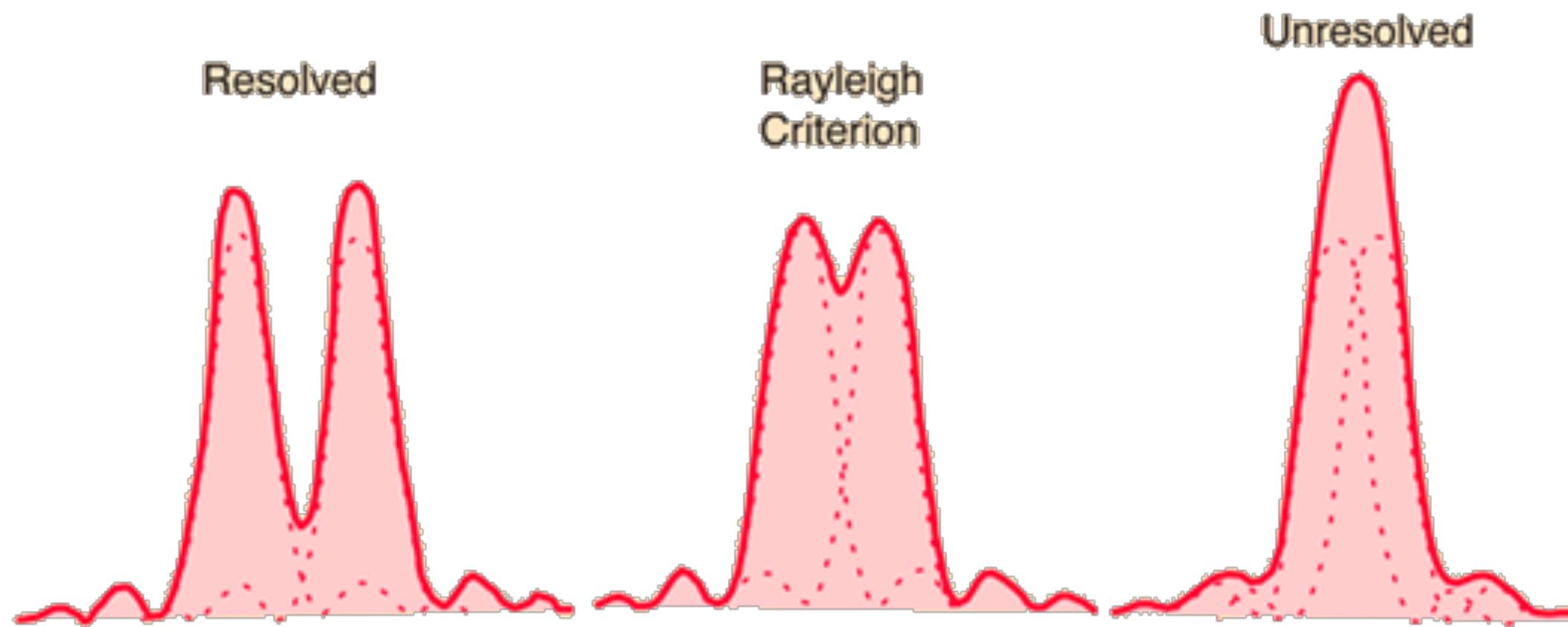
# Resolution limit of an optical microscope system

Rayleigh criterion:

$$\sin \theta = 1.22 \frac{\lambda}{D}$$

or

$$\Delta l = 1.22 \frac{f\lambda}{D}$$



$\theta$  is the angular resolution,  $\lambda$  is the wavelength and  $D$  is the diameter of the lens aperture.

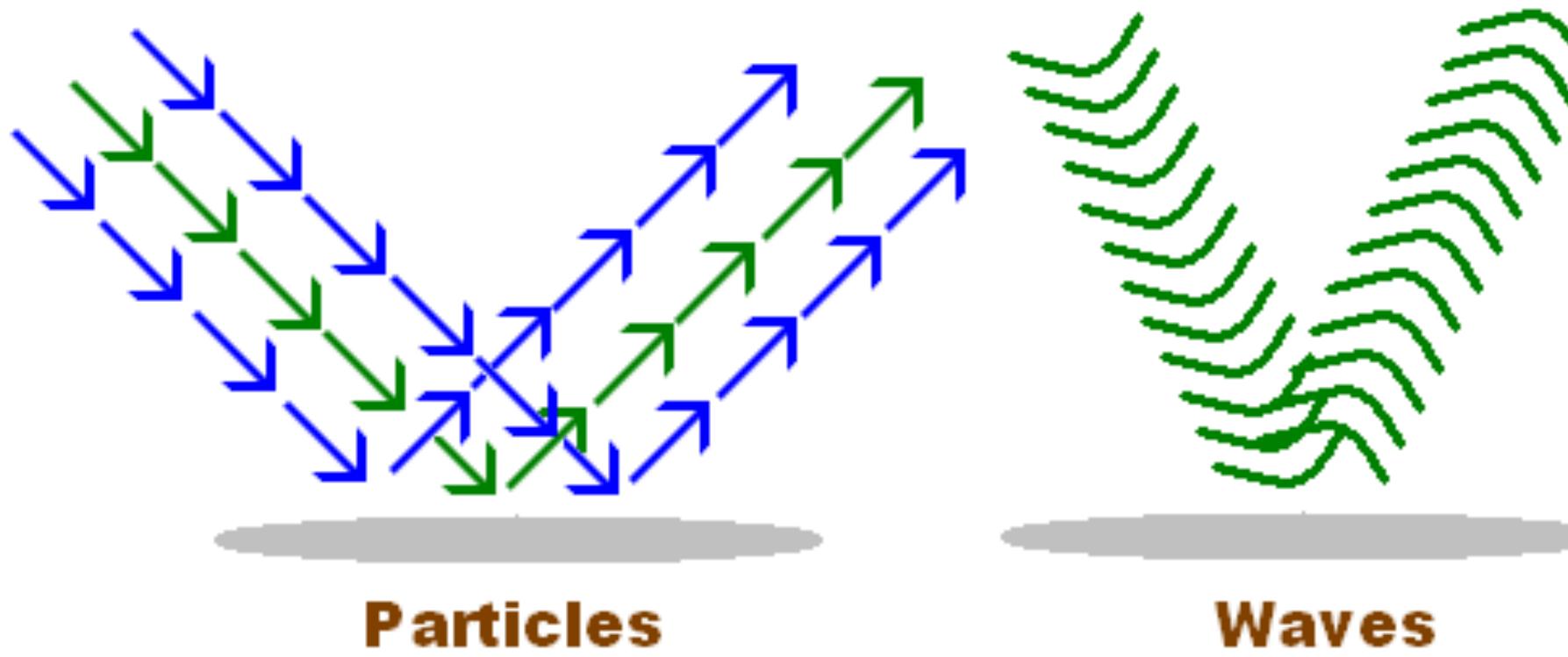
or:  $\Delta l$  is the spatial resolution,  $f$  is the focal length of an ideal lens.

Thus: The resolution of an light microscope system is limited by the wavelength of the light used. One of the ways to improve the resolution is to use light with shorter wavelength or use larger lens aperture.

# Wave-particle duality of electron

It all started with the De Broglie's hypothesis:

$$\lambda = \frac{h}{p}$$



$\lambda$  is wavelength,  $h$  is Planck's constant, and  $p$  is momentum.

The original motivation of building an electron microscope came from the shorter wavelength of the electron.

# Electron wavelength

Applying the principle of energy conservation to an electron (-e) traveled in voltage  $E_0$ :

$$eE_0 = \frac{h^2}{2m\lambda^2}$$

$$\lambda = \frac{h}{\sqrt{2meE_0}}$$

$E_0$  = acceleration voltage

$\lambda$  = wavelength

$h$  = Planck's constant

$m$  = electron mass

$e$  = electron charge

## Electron wavelength

Take the relativity into consideration, the wave length is:

$$\lambda = \frac{h}{\sqrt{2m_0eE_r}} \quad E_r = E_0 + \left( \frac{e}{2m_0c^2} \right) E_0^2$$

$$\lambda = \frac{1.22639}{\sqrt{E_0 + 0.97845 \times 10^6 E_0^2}}$$

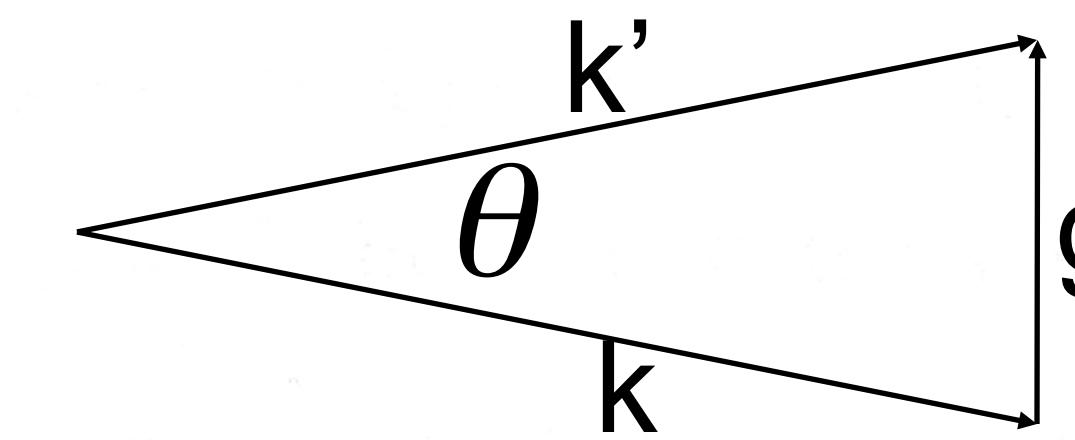
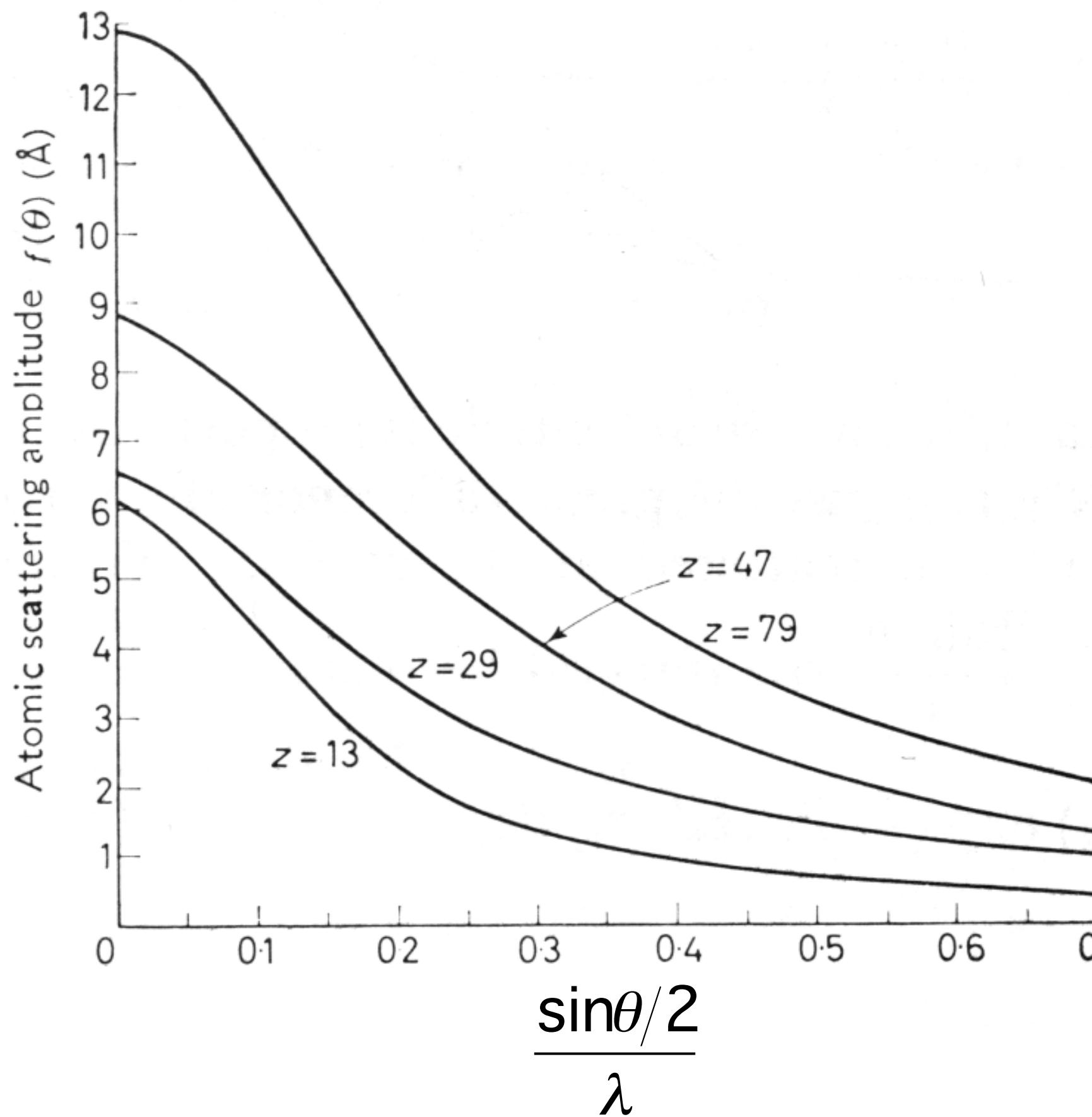
120kV  $\lambda=0.033\text{\AA}$ ; 200kV  $\lambda=0.025\text{\AA}$ ; 300kV  $\lambda=0.020\text{\AA}$ ;

Note that these wavelength is considerably shorter than that used in X-ray crystallography,  $\sim\text{\AA}$ .

# Atomic Scattering Factor for Electrons

Mott formula:

$$f_e(\theta) = \frac{m\epsilon^2}{2h^2} \left( \frac{\lambda}{\sin\theta/2} \right)^2 [Z - f_x(\theta)]$$



$$|\vec{g}| = \frac{2\sin\frac{\theta}{2}}{\lambda} = 2|\vec{k}|\sin\frac{\theta}{2}$$

Figure 4.6. Atomic scattering amplitudes as a function of  $\sin\frac{1}{2}\theta/\lambda$  for Al ( $Z=13$ ), Cu ( $Z=29$ ), Ag ( $Z=47$ ) and Au ( $Z=79$ )

# **Electron v.s. X-ray**

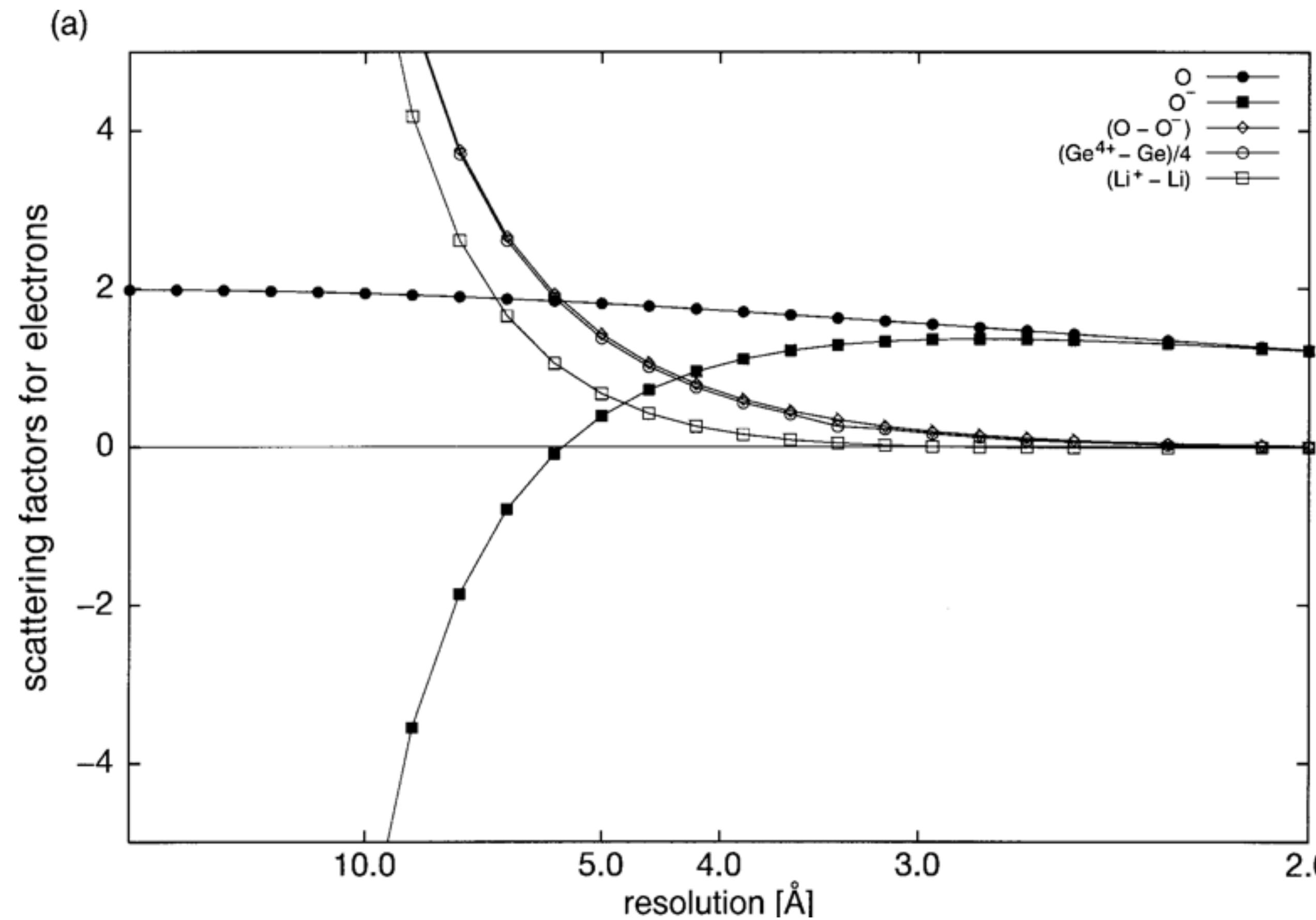
As particles:

- Electrons interact with the potential field of an atom, including shell electrons and nucleus, X-rays interact with only shell electrons;
- Electrons have much larger scattering cross-sections than X-rays; multiple scattering is severer in electron scatterings than in X-ray diffraction; For biological sample, radiation damage is also severer than X-ray diffraction.

As wave:

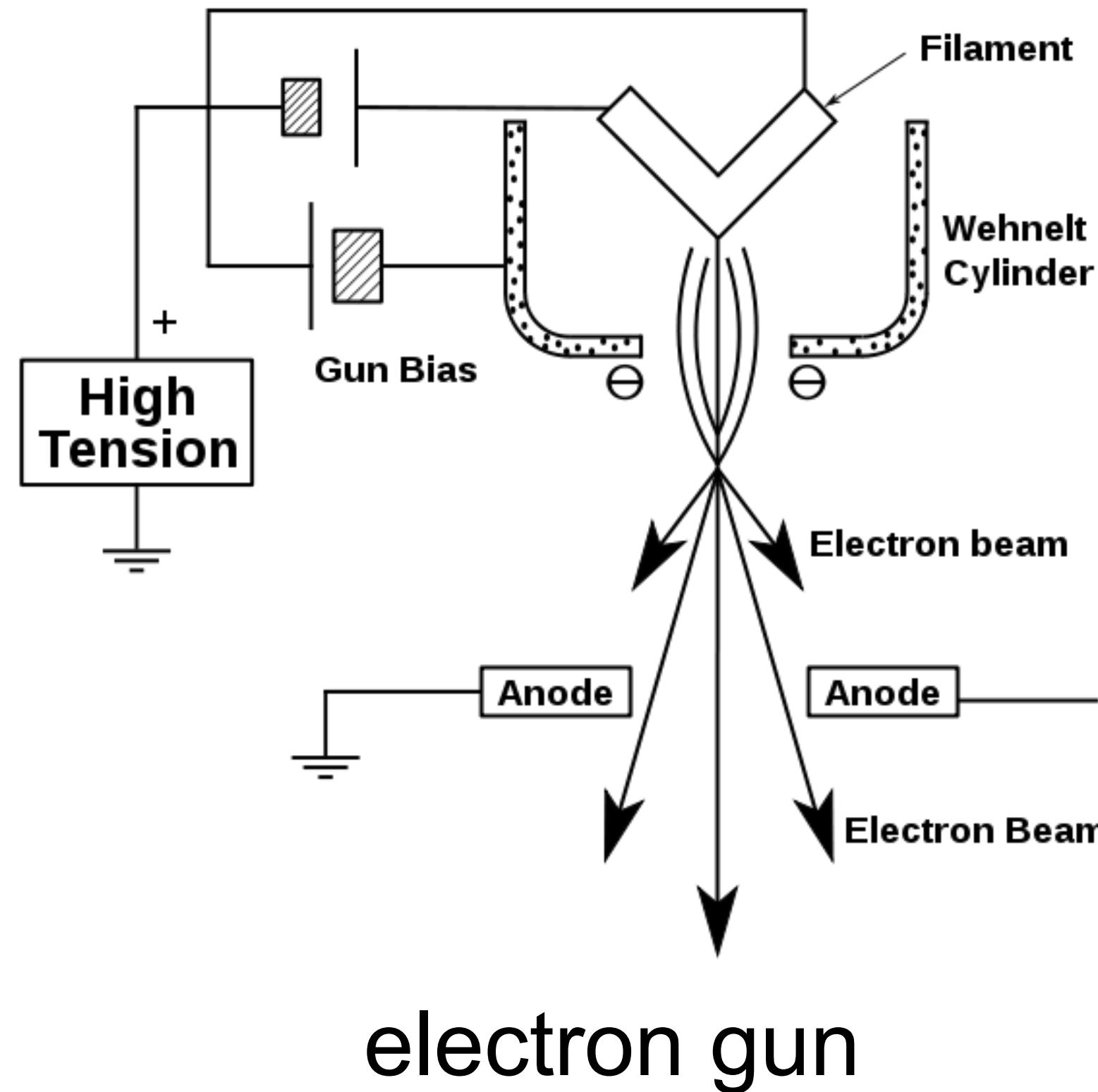
- Electrons can be focused by electromagnetic lens, X-ray can not be focused by lens;

# Electron v.s. X-ray

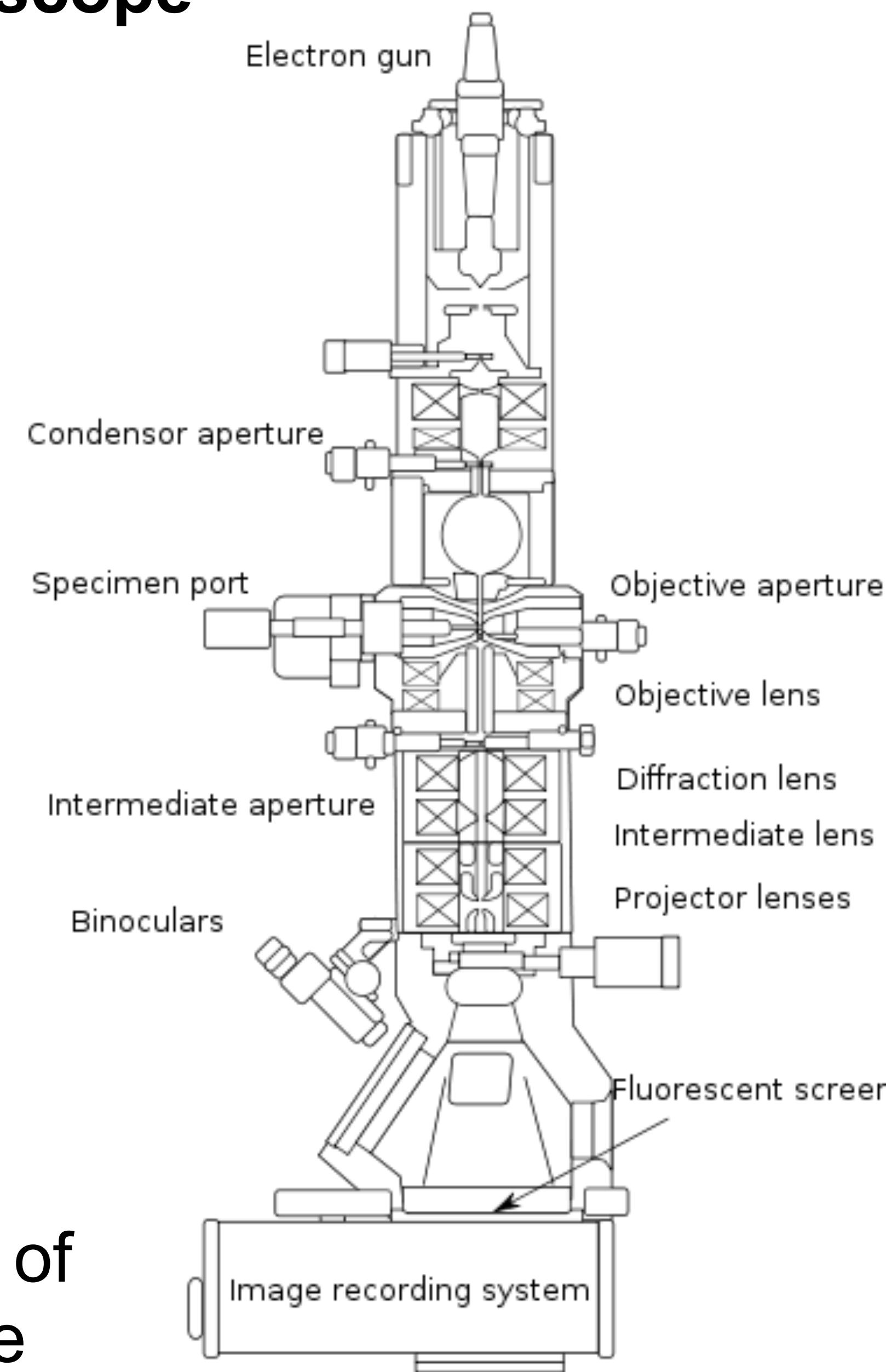


Mitsuoka et al. (1999) JMB, **286**, 861-882.

# Electron Microscope



a schematic drawing of  
electron microscope



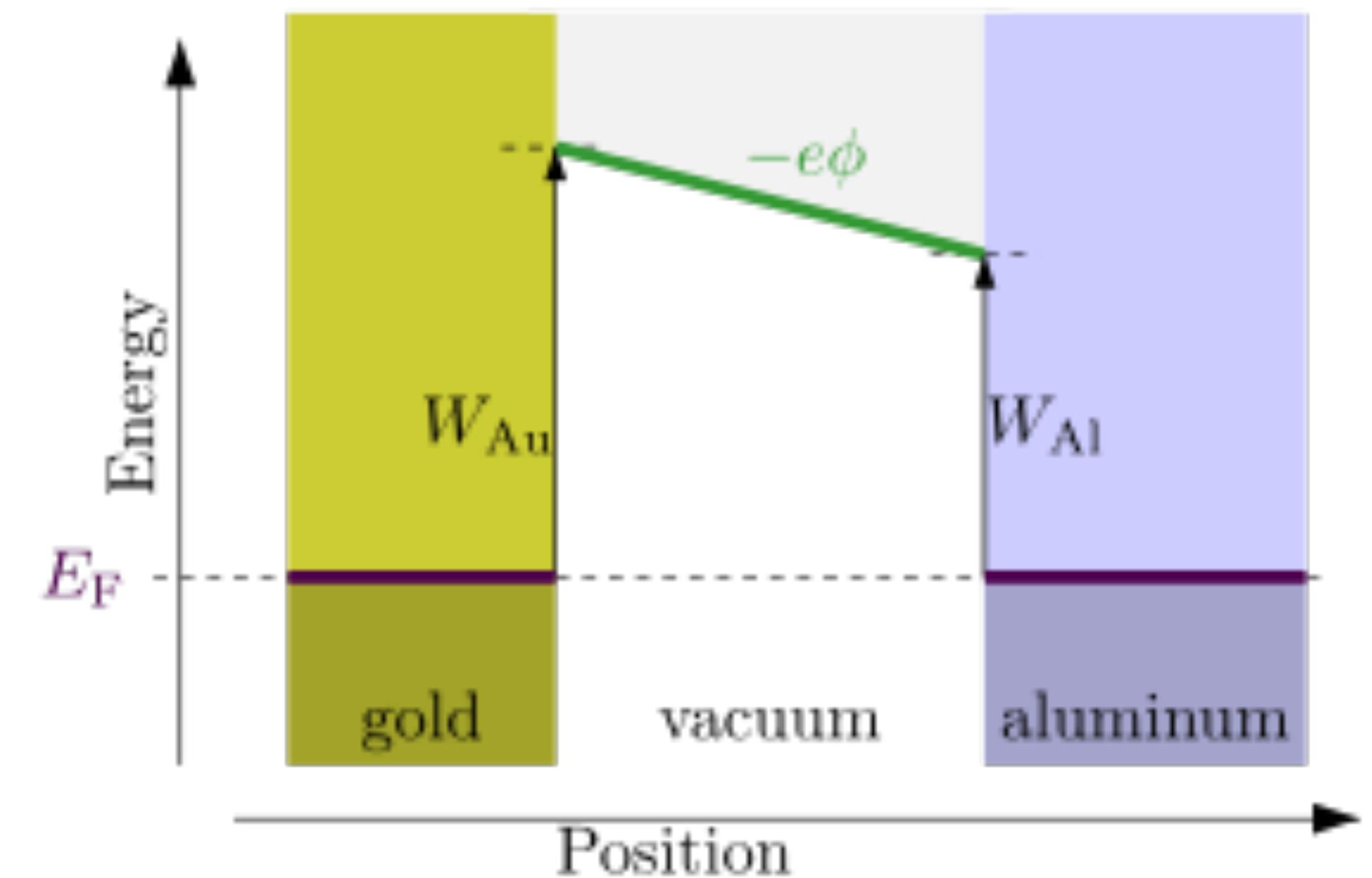
# Work-function

**Work function:** minimum thermodynamic energy (work) needed to remove an electron from a solid to a point in vacuum immediately outside the solid surface.

$$W = -e\phi - E_F$$

W: work function, -e charge of electron,  $\phi$  electron static potential in vacuum, and EF is the Fermi level of the materials.

Work function is a fixed character of material. Work function of tungsten is 4.50eV.



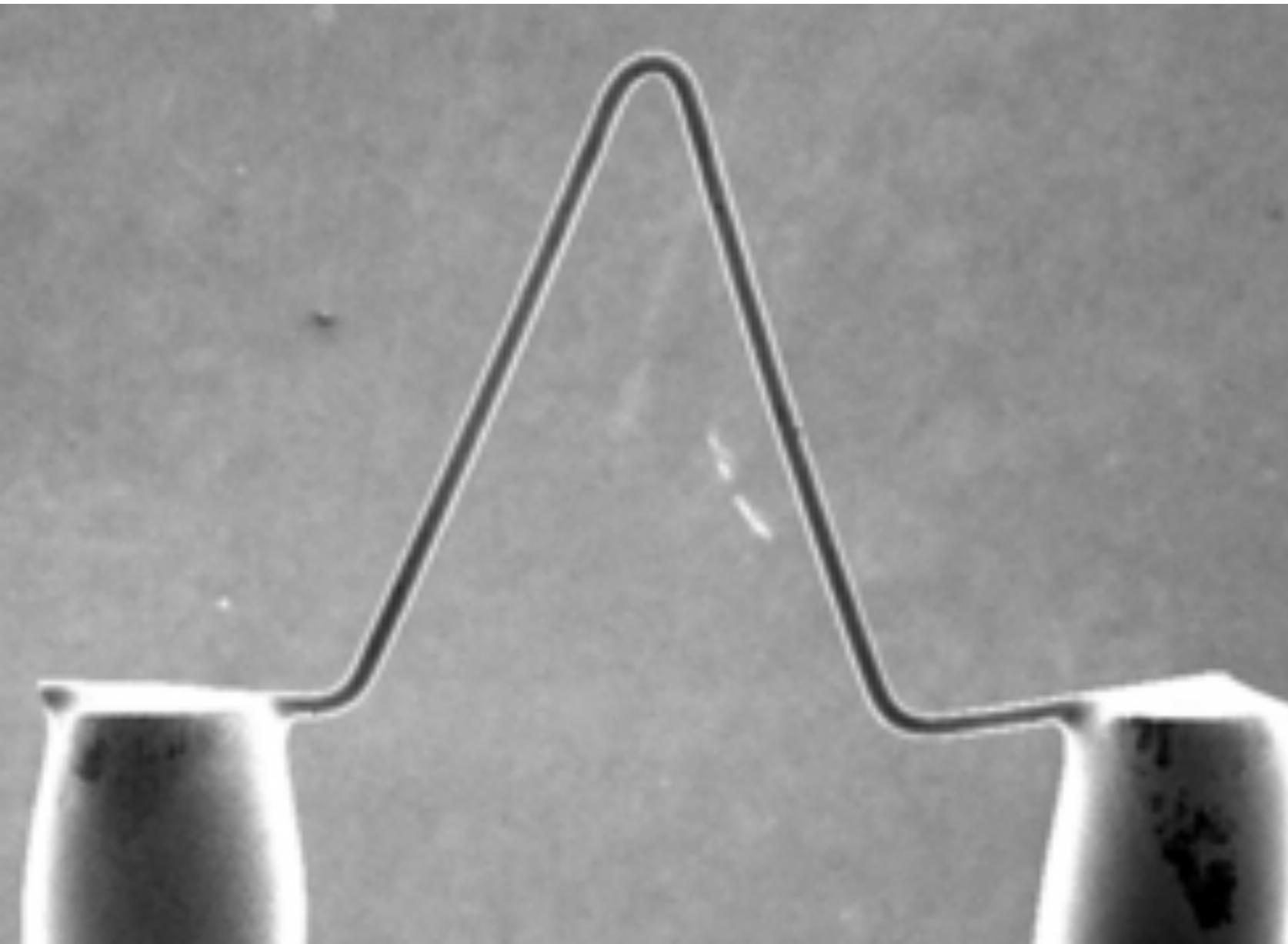
electron emission from electron gun: giving sufficient energy to electron so that it can escape from material's surface.

# Electron source

**Electron source:** require enough energy to overcome “work function” of materials to escape from its surface into vacuum;

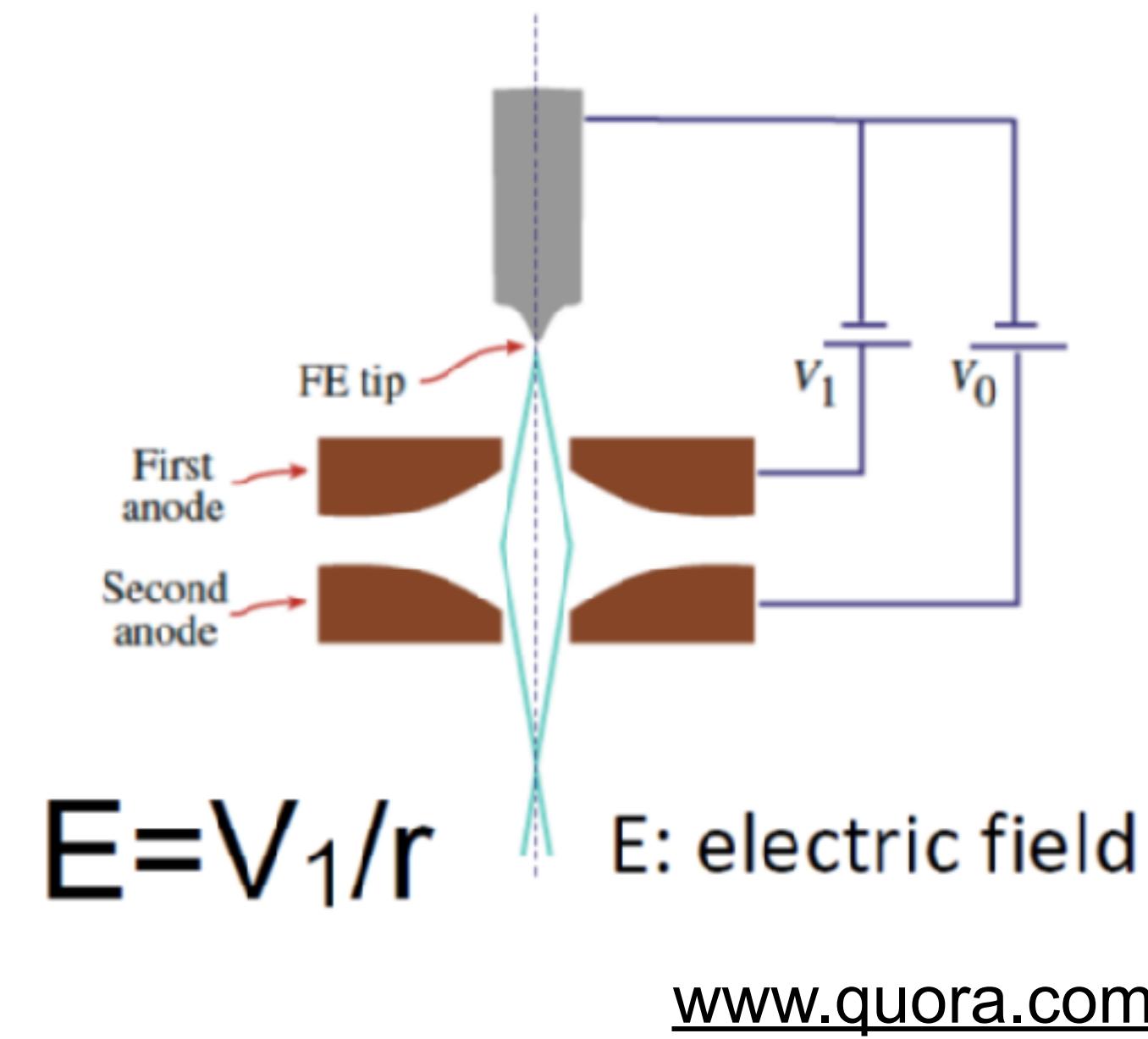
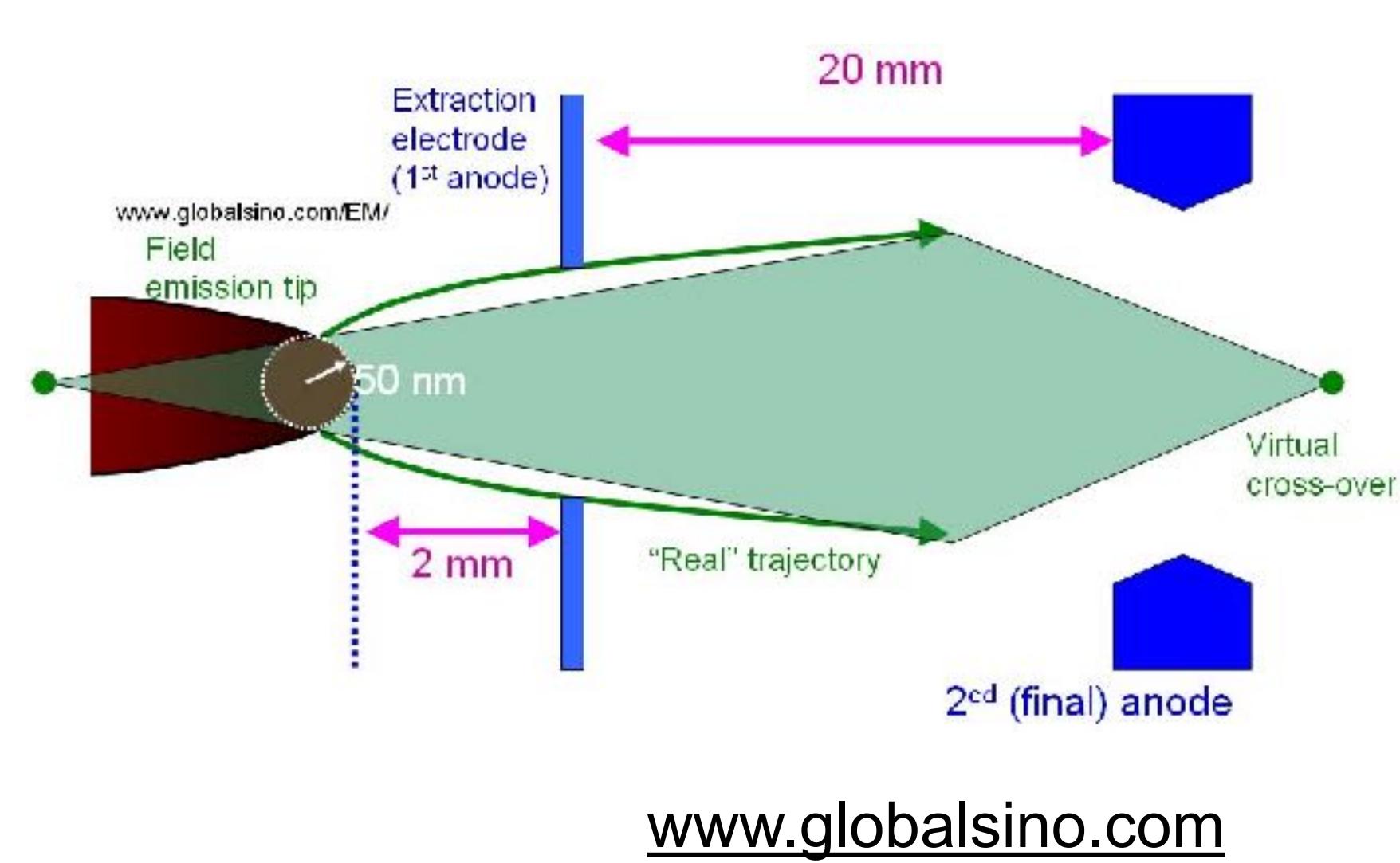
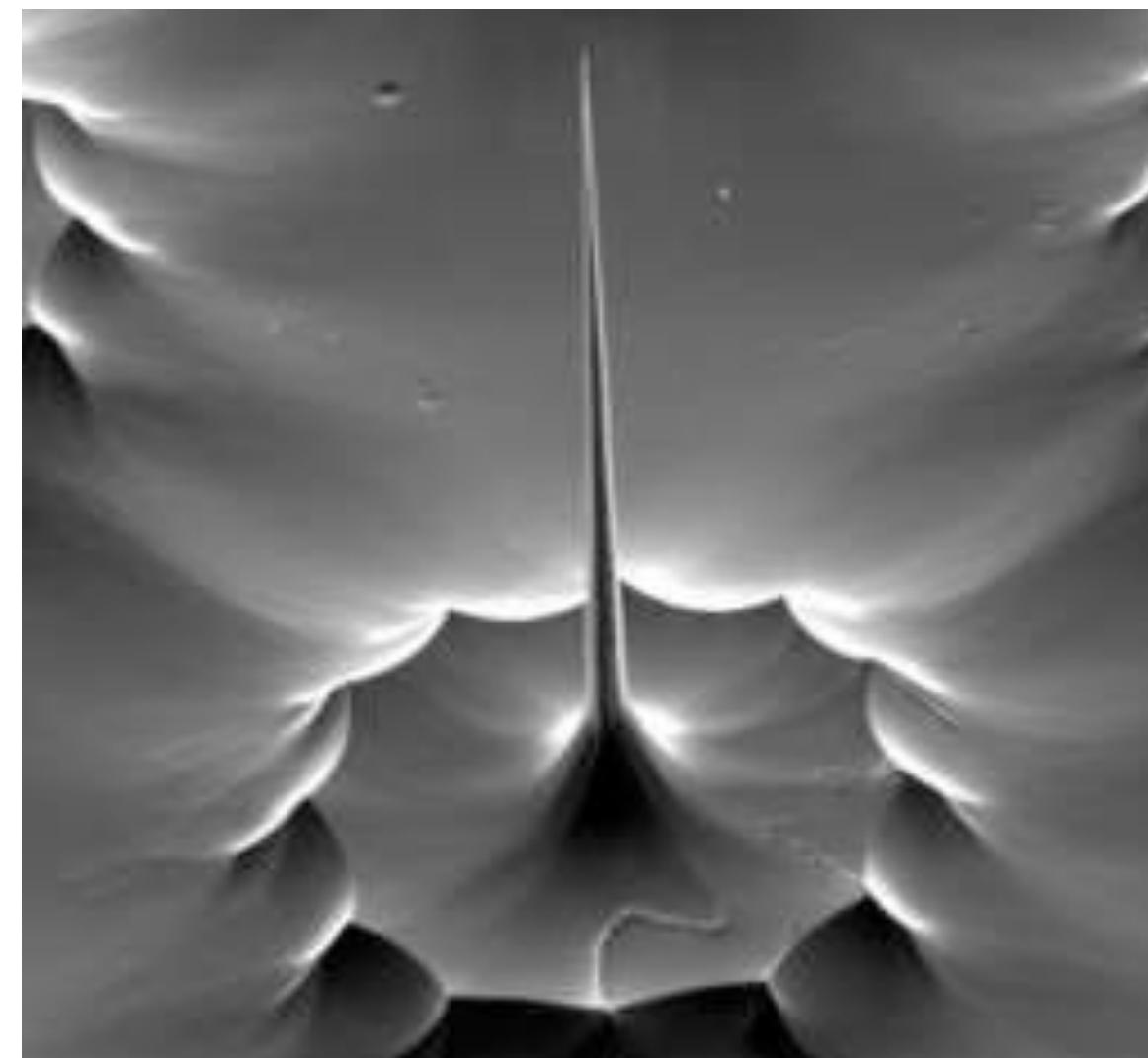
**Thermo ionic source:** thermal energy induced flow of electron from a surface.

- \* electron beam coherence depends on the size of the tip - smaller the tip size, better the coherence;
- \* tungsten filament is larger than LaB<sub>6</sub> crystal



# Field emission gun (FEG)

FEG tip is very sharp. Applying a few kV potential generates a sufficient potential gradient at the tip so that electron can acquire enough energy to overcome “work function” of materials and to escape from its surface into vacuum;

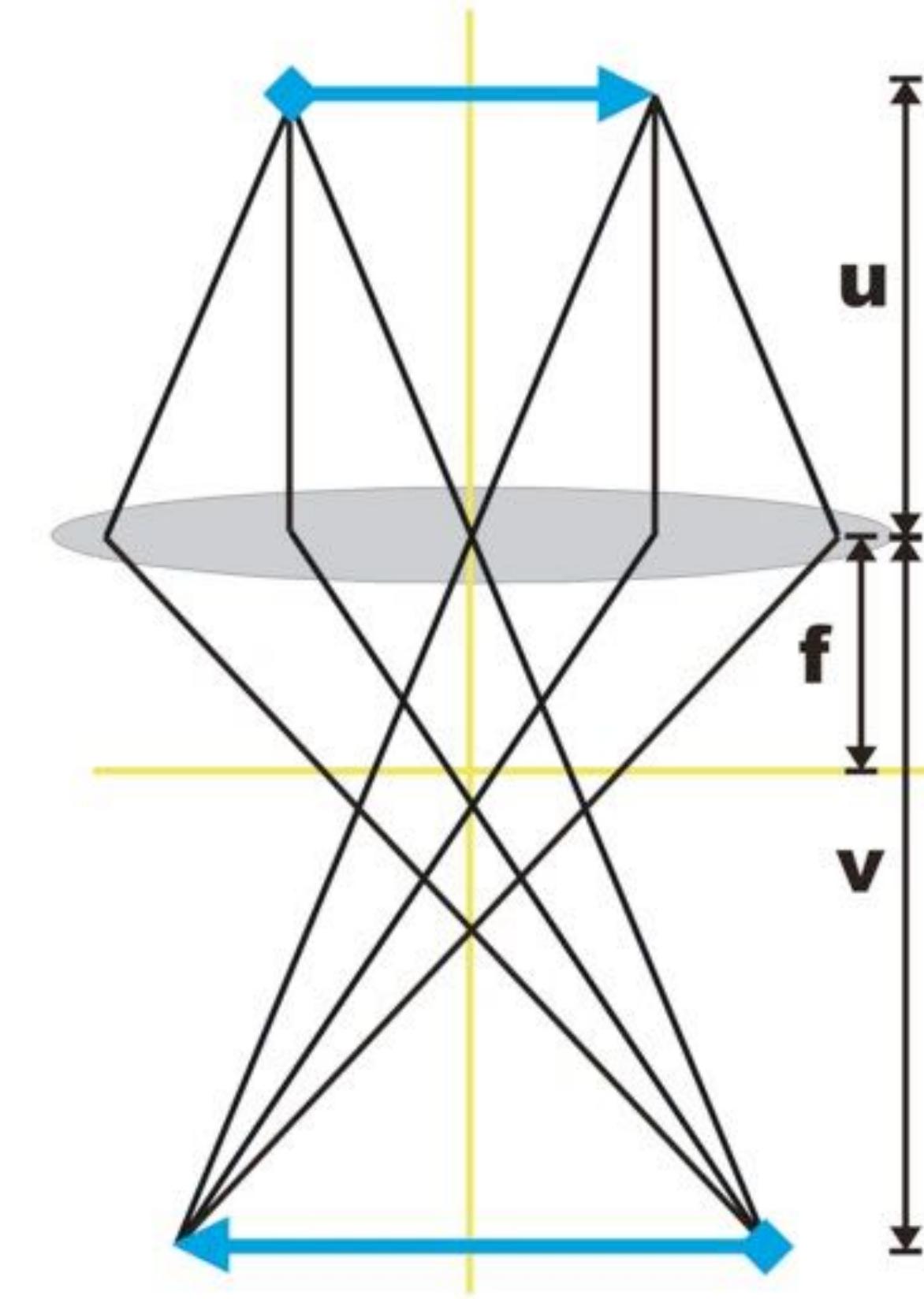
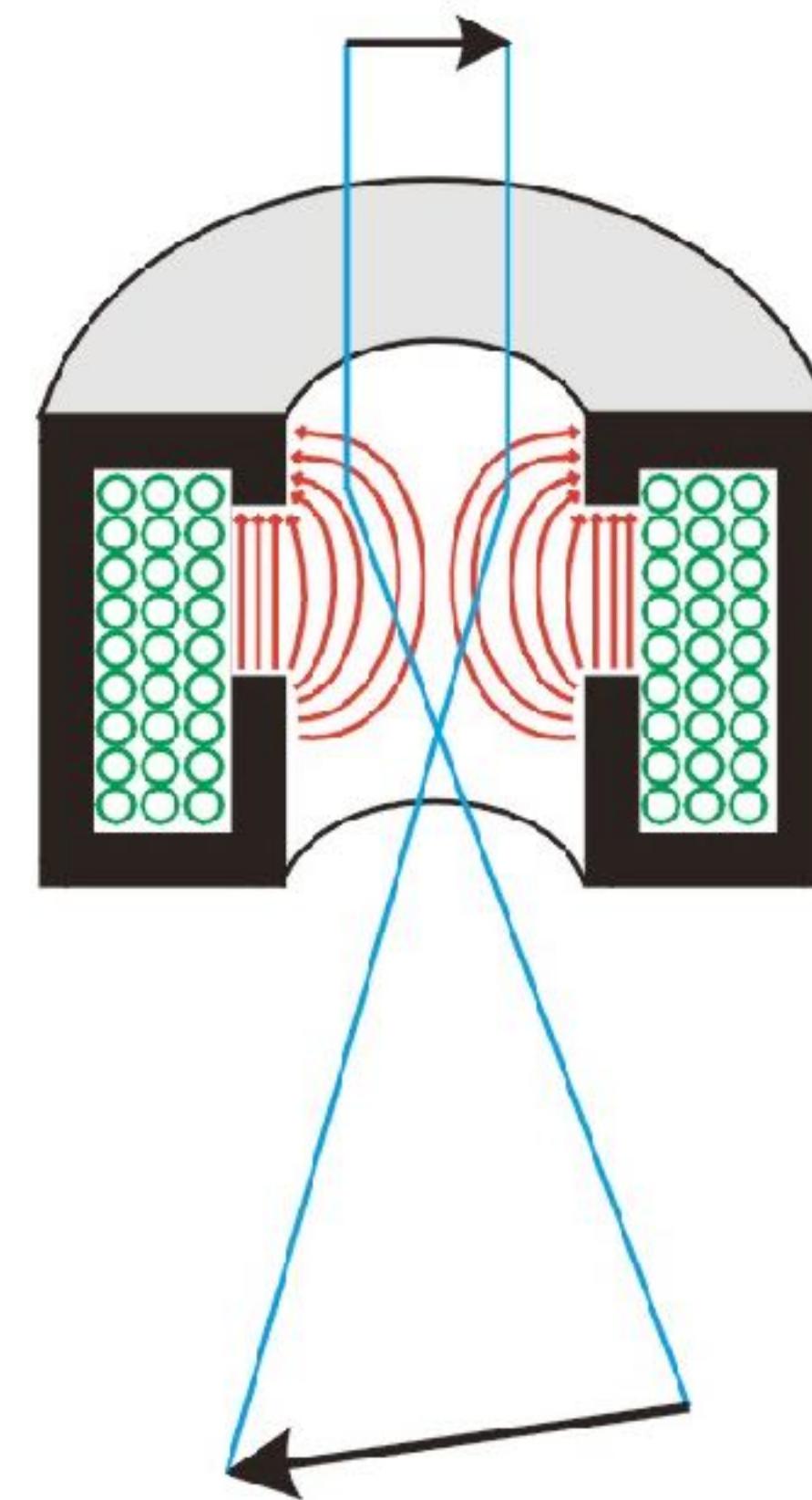


**Cold emission:** single crystal tungsten, tip size smaller than 100nm; It has the highest coherence. It requires high vacuum to operate; contamination on surface reduces the emission, requires frequent flash; energy spread is  $\sim 0.4\text{eV}$

**Schottky tip:** tungsten sharp tip coated with ZrO. Operates at a higher temperature to facilitate electron emission, tip size larger than cold tip, but requirement for vacuum is less, and stable. Energy spread is  $\sim 0.7\text{eV}$

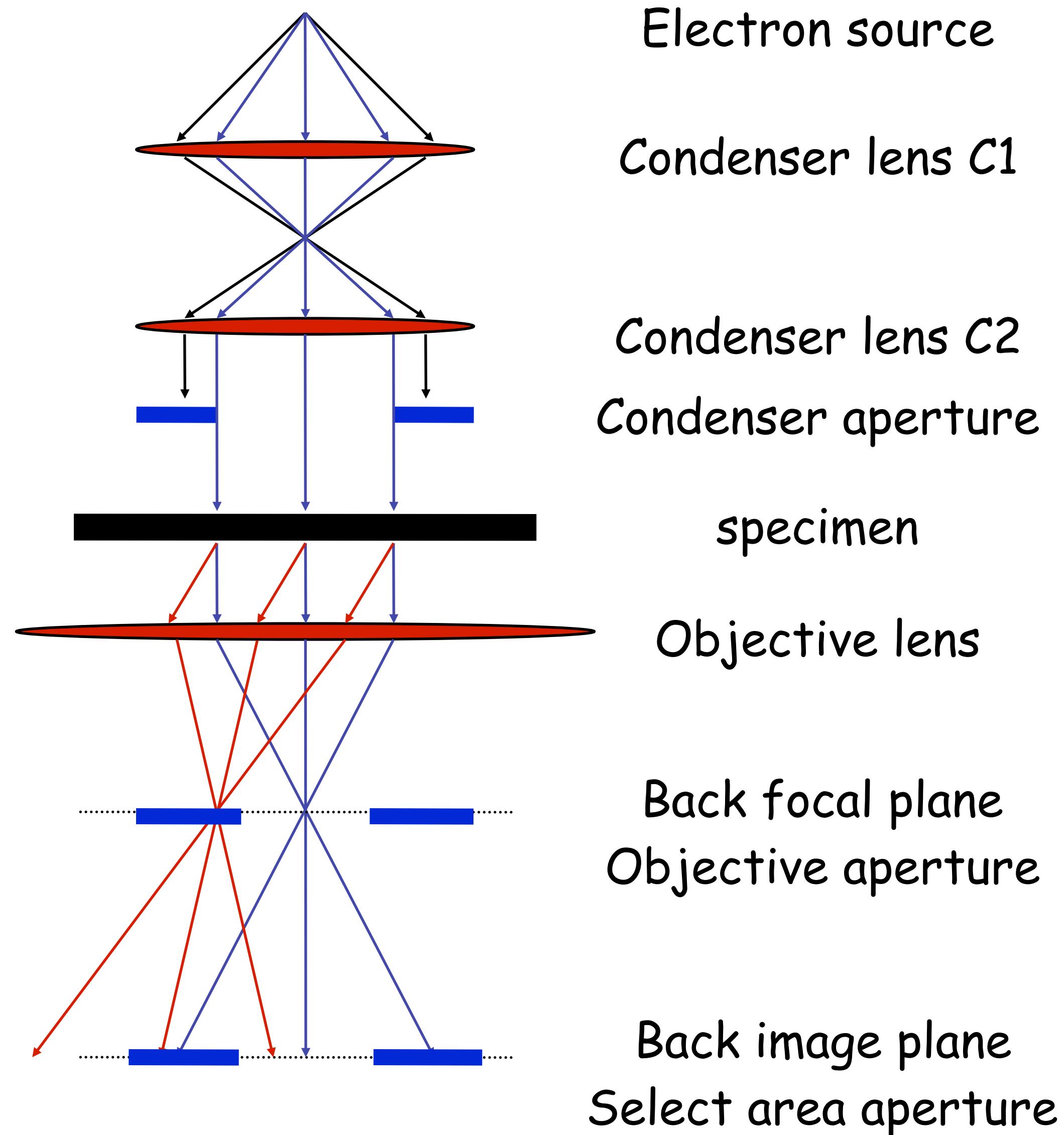
**Monochrometer:** energy spread  $\sim 0.1\text{eV}$

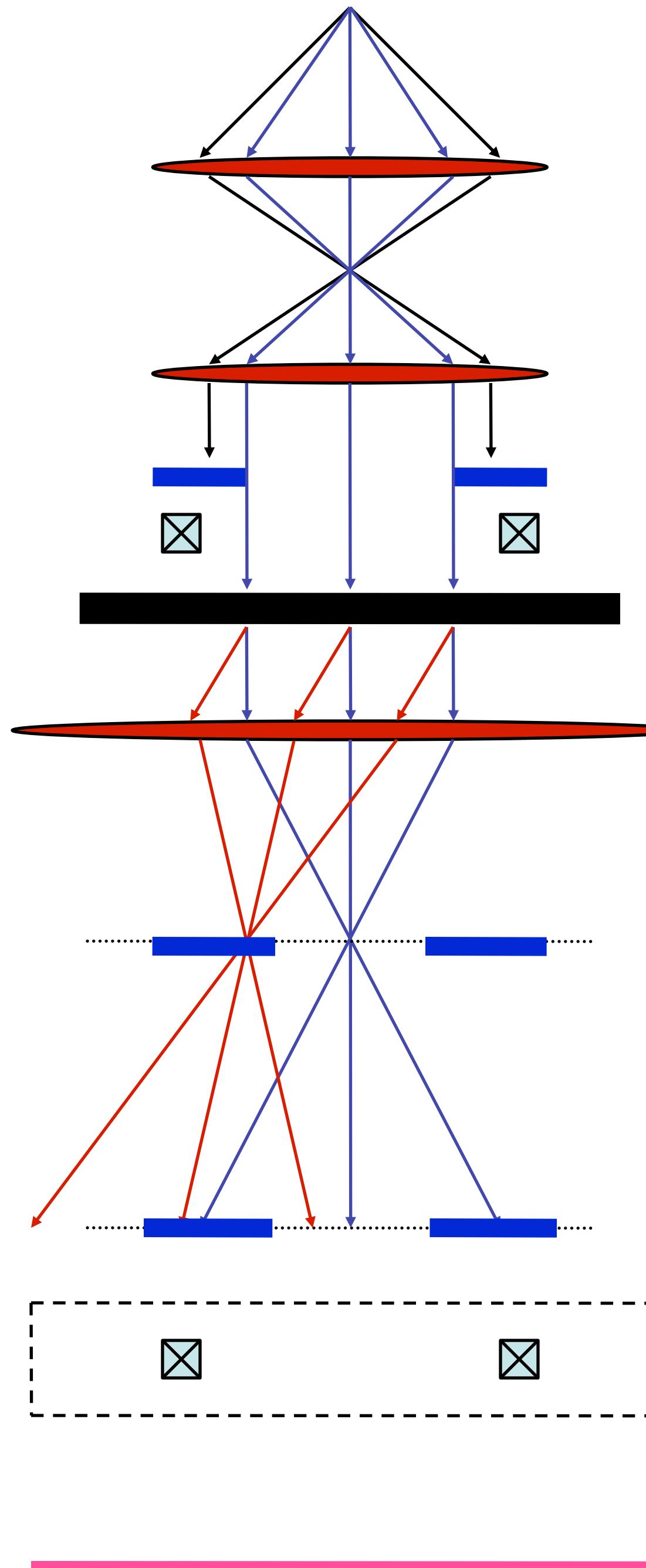
# Electromagnetic lens



\* The focal length of a electromagnetic lens can be easily adjusted by changing the lens current.

# Optic system in an electron microscope



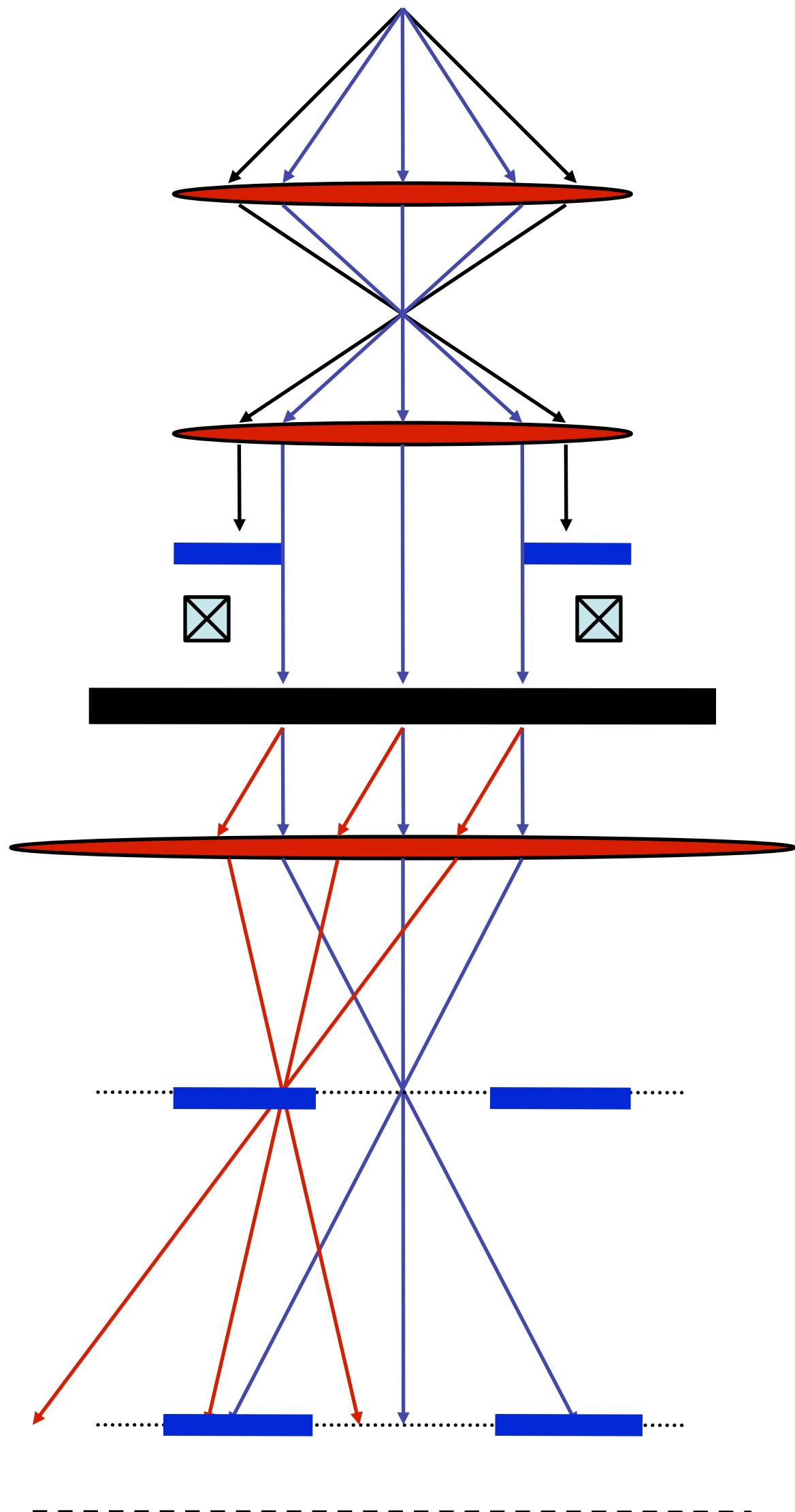


## Additional lens in the electron microscope

Beam shift coils

Projection lens system  
Image shift/diffraction shift coils

screen



## Image mode

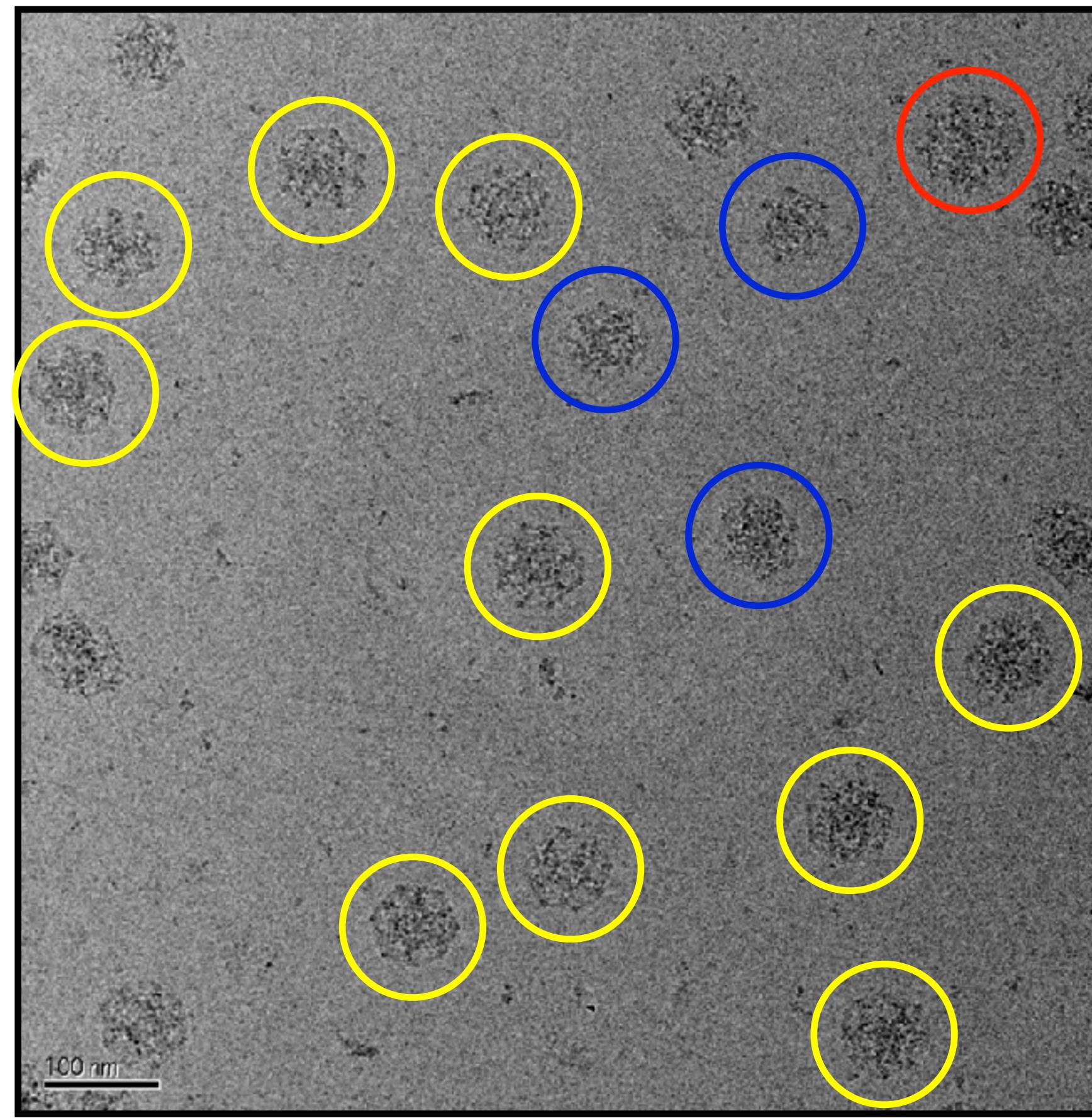
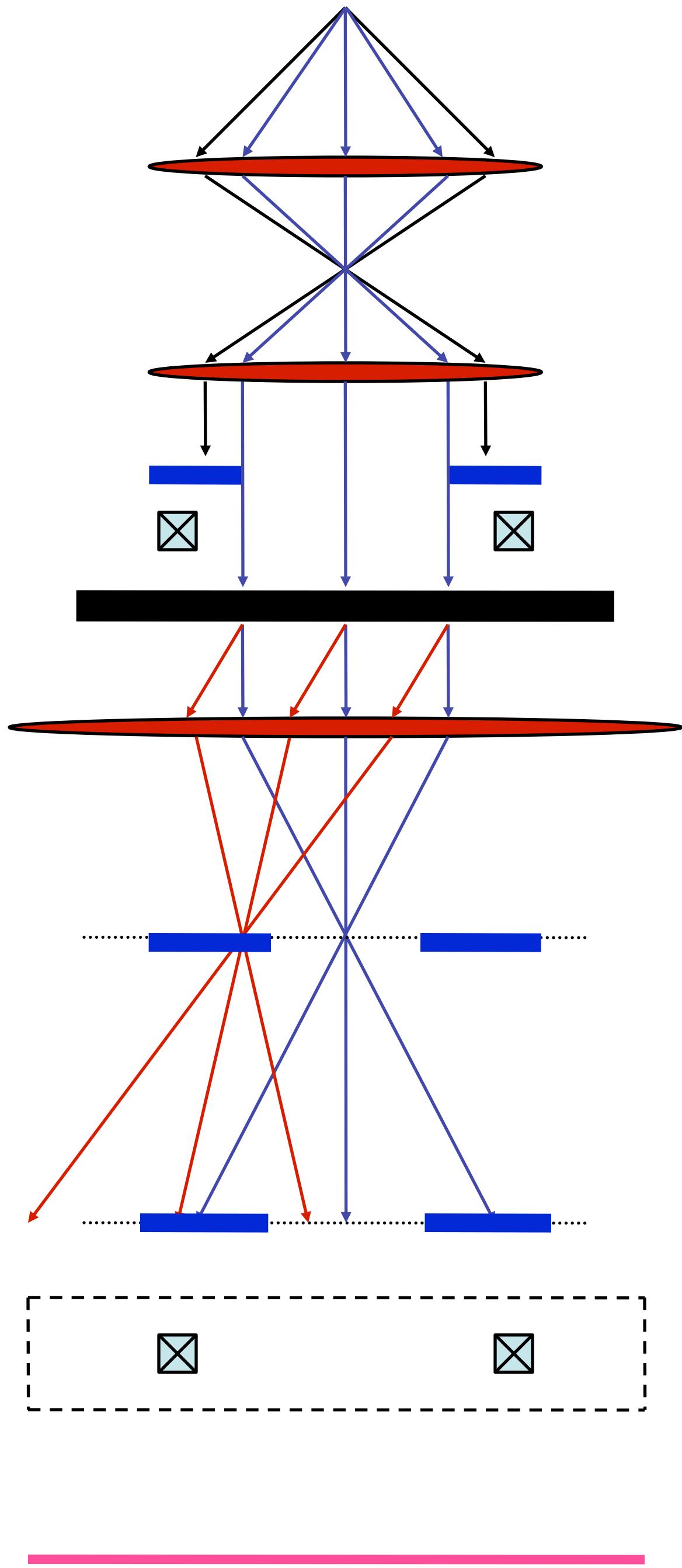
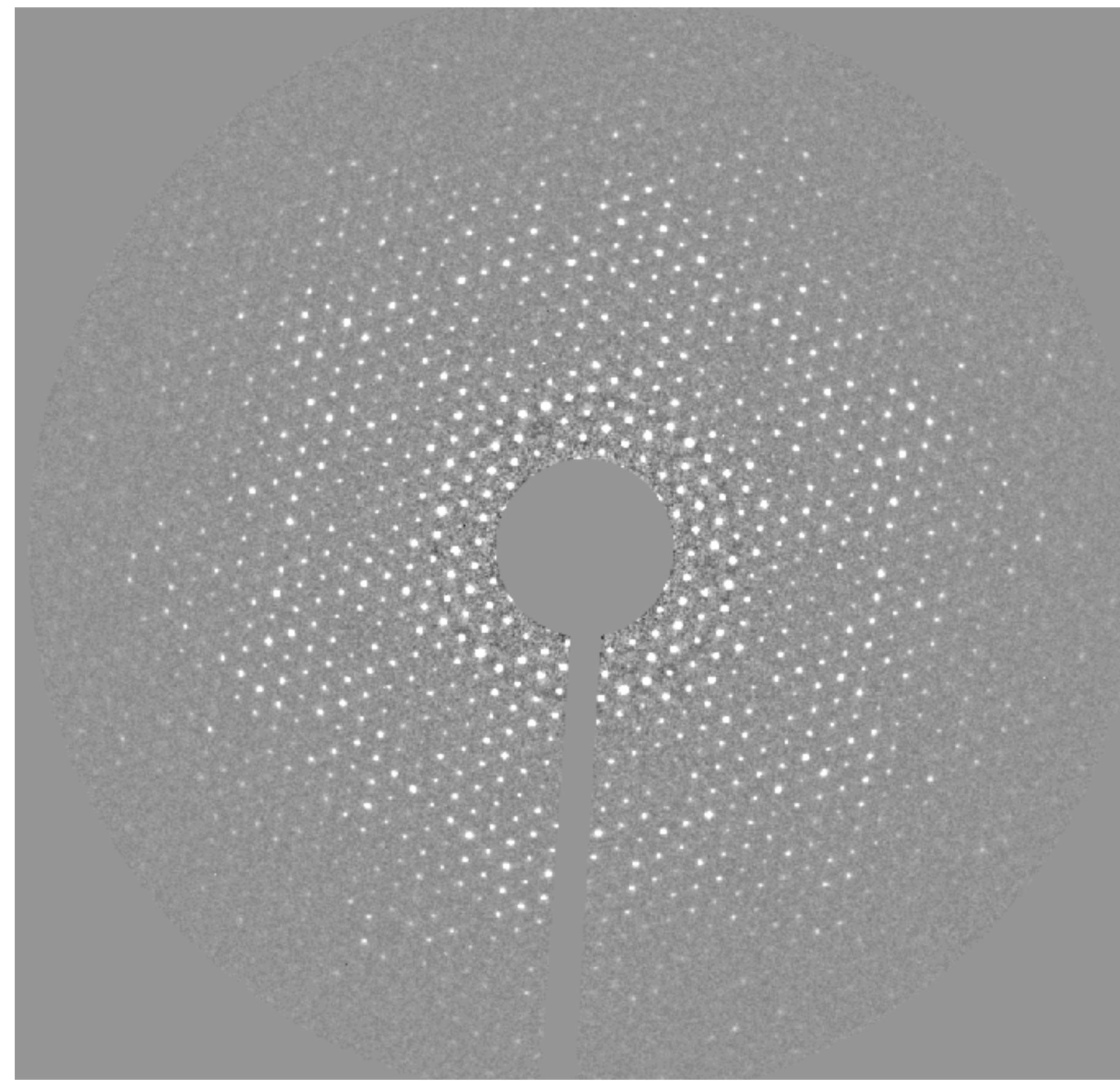


Image mode

Clathrin coat



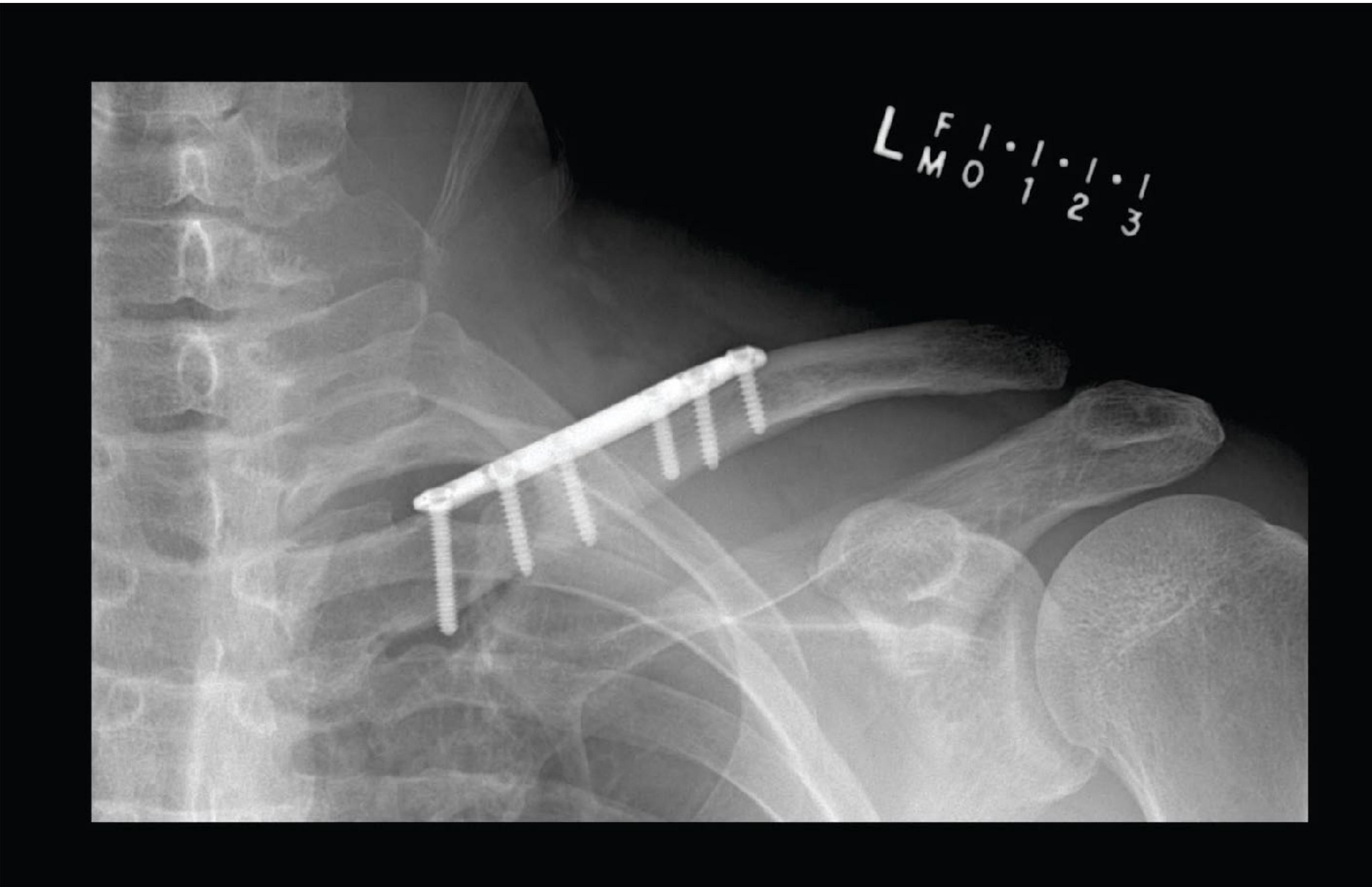
## Diffraction mode



Diffraction mode

bacteriorhodopsin

# EM images are projections



# Image formation

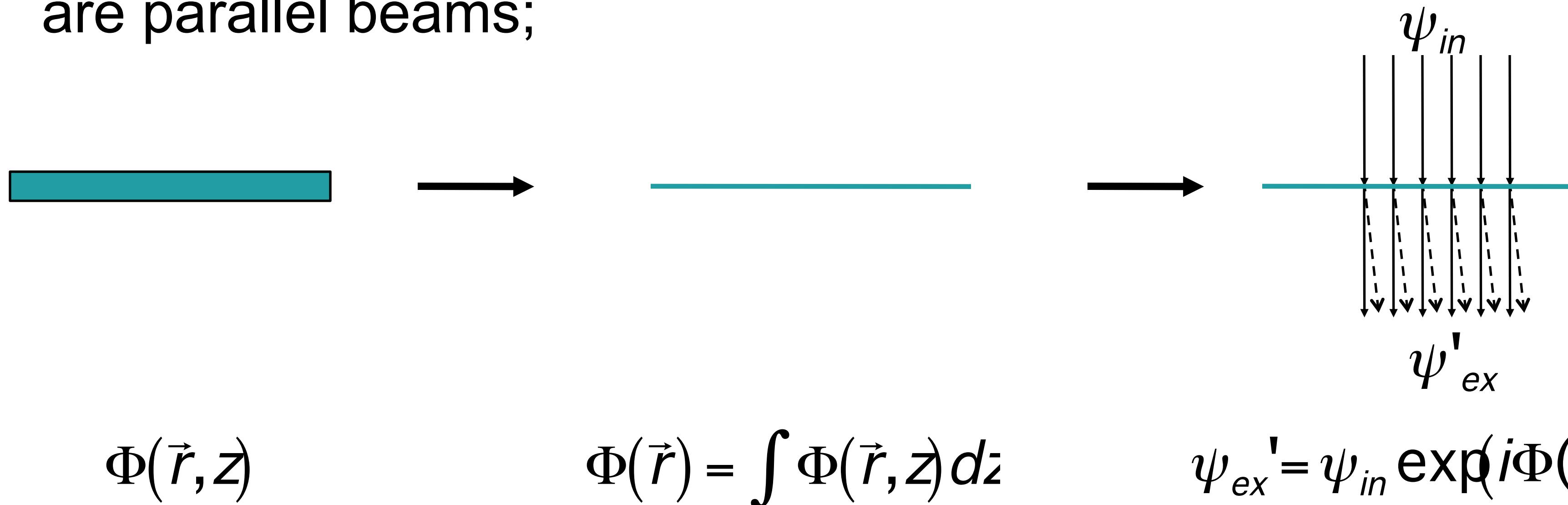
The image formation in the electron microscope can be treated as two separate processes:

- 1) The interaction of the incident beam with the specimen, described by the weak-phase object approximation, which is the theory used mostly to describe the image formation of thin specimen with light elements, such as a biological sample.
- 2) The propagation of the electron beam from exit plane of the specimen to the back image plane of the focus lens.

## Weak-phase object approximation

This is a highly simplified theory based on the so-call weak-phase object, which is a very thin specimen formed mostly by low- and medium-weight molecules.

Suppose: 1) the specimen is very thin so that  $\Phi(\vec{r}, z)$  can be approximated by  $\Phi(\vec{r})$ ; 2) both in-coming and exiting beams are parallel beams;



$$\psi_{ex}' = \psi_{in} \exp(i\Phi(\vec{r})) \quad (2)$$

$$\psi_{ex}' = \psi_{in} \left[ 1 + i\Phi(\vec{r}) - \frac{1}{2}\Phi^2(\vec{r}) + \dots \right] \quad (3)$$

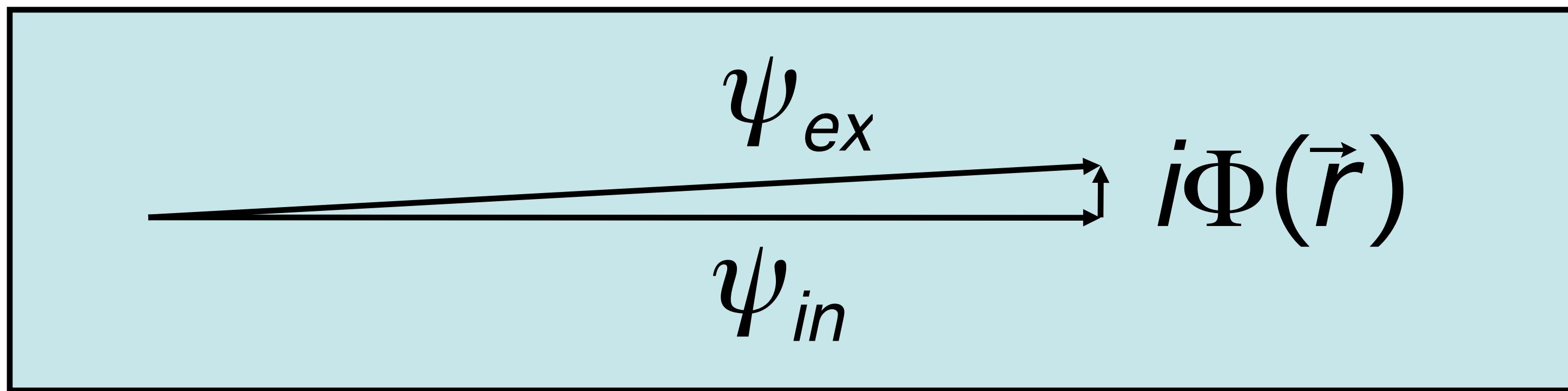
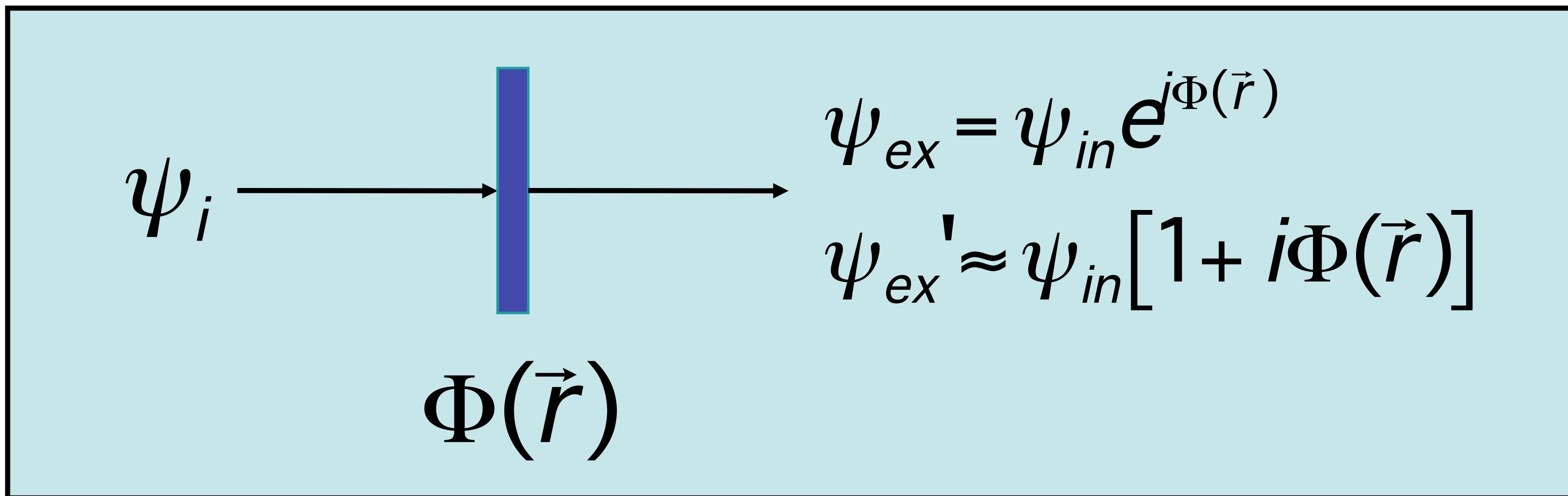
The first term in (3) represents the central unscattered beam, the second term the kinematically scattered beam and the higher terms are for the dynamical scattering. The weak phase object approximation assumes that  $\Phi(r) \ll 1$ , the phase shift is so small that the following approximation will work:

$$\psi_{ex}' \approx \psi_{in} [1 + i\Phi(\vec{r})] \quad (4)$$

Taking absorption into consideration:

$$\psi_{ex}' = \psi_{in} \exp(i\Phi(\vec{r}) - \mu(\vec{r})) \quad (5)$$

$$\psi_{ex}' \approx \psi_{in} [1 - \mu(\vec{r}) + i\Phi(\vec{r})] \quad (6)$$



$$I_{ex} = |\psi_{ex}|^2 = |\psi_{in}|^2 = I_{in}$$

# Image formation

At exit plane of specimen:

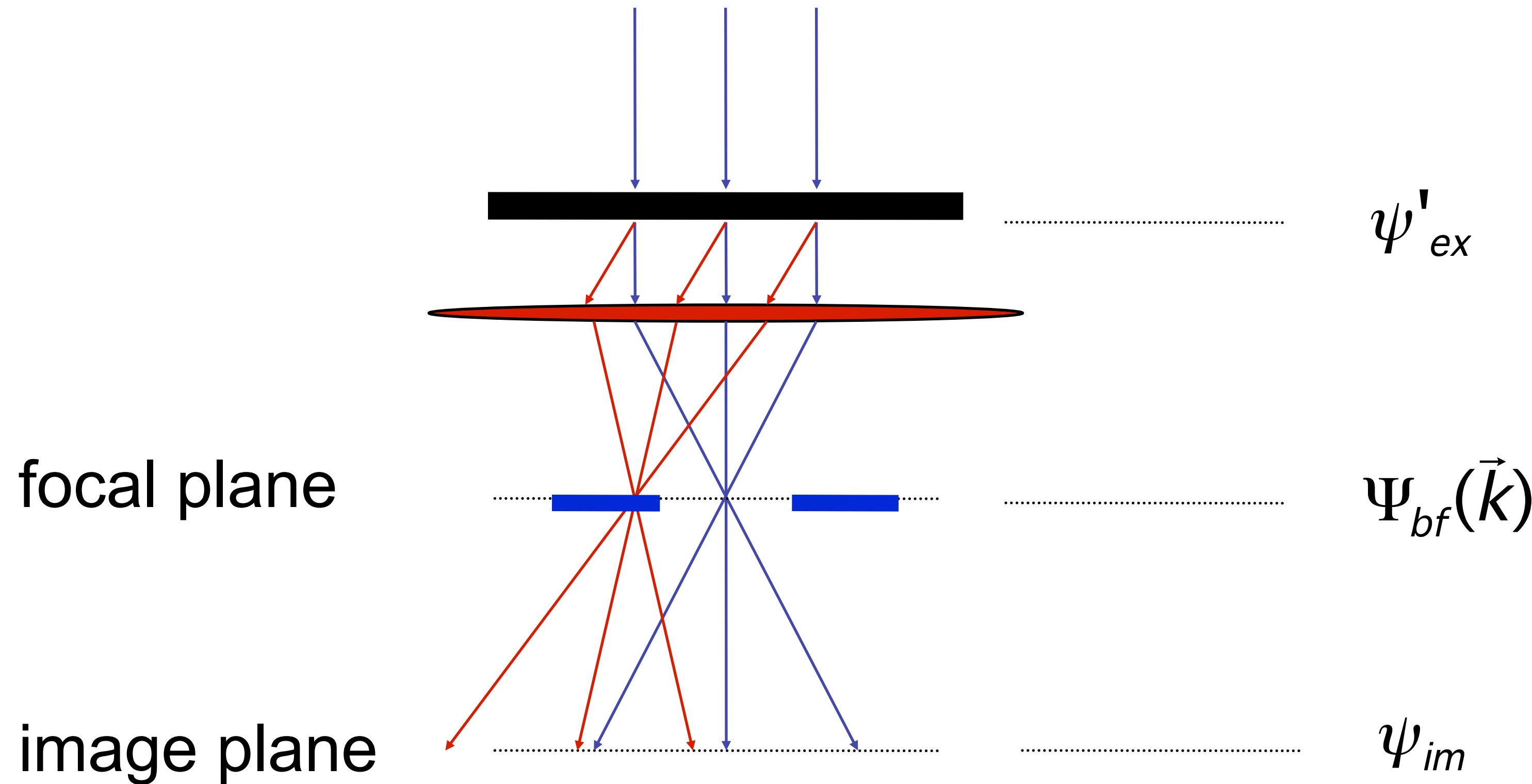
$$\psi'_{ex}(\vec{r}) \approx \psi_{in}[1 + i\Phi(\vec{r})]$$

At back focal plane:

$$\Psi_{bf}(\vec{k}) = F[\psi'_{ex}(\vec{r})]$$

At back image plane:

$$\psi_{im}(\vec{r}) = F^{-1}[\Psi_{bf}(\vec{k})]$$



## Image formation

The plane wave  $\psi'$  of exit-beam travel through objective lens to the back focal plane. The wave function at back focal plane of the objective lens is the Fourier transform of the exit wave:

$$\Psi_{bf}(\vec{k}) = F(\psi'_{ex}(\vec{r})) = F[1 + i\Phi(\vec{r})] = \delta(\vec{k}) + iF(\Phi(\vec{r})) \quad (7)$$

However the lens aberration and defocusing generate an extra phase shift to the scattered beam:

$$\begin{aligned} \gamma(\vec{k}) &= 2\pi\chi\vec{k} \\ \chi(k, \varphi) &= \frac{1}{2}\lambda\left[\Delta z + \frac{1}{2}\sin 2(\varphi - \varphi_0)\right]k^2 + \frac{1}{2}\lambda^3 C_s k^4 \end{aligned} \quad (8)$$

Together with the aperture function  $A(\vec{k})$  the wave function at back focal plane will become:

$$\Psi_{bf}(\vec{k}) = F(\psi'_{ex}) A(\vec{k}) \exp(\varrho \pi i \chi \vec{k}) \quad (9)$$

Then, the wave function in the back image plane of the lens is the reverse Fourier transform of the wave function at back focal plane ( $\otimes$  is for convolution):

$$\begin{aligned} \psi_{im}(\vec{r}) &= F^{-1}\{F(\psi'_{ex}) A(\vec{k}) \exp(\varrho \pi i \chi \vec{k})\} \\ &= 1 + i\Phi(-\vec{r}) \otimes J_0(\vec{r}) \otimes F^{-1}[\exp(\varrho \pi i \chi \vec{k})] \end{aligned} \quad (10)$$

The observed intensity in the image is then:

$$\begin{aligned} I_i(\vec{r}) &= \psi_i(\vec{r}) \psi_i^* \\ &= 1 + 2\Phi(-\vec{r}) \otimes J_0(\vec{r}) \otimes F^{-1}[\sin(\varrho \pi \chi \vec{k})] \end{aligned} \quad (11)$$

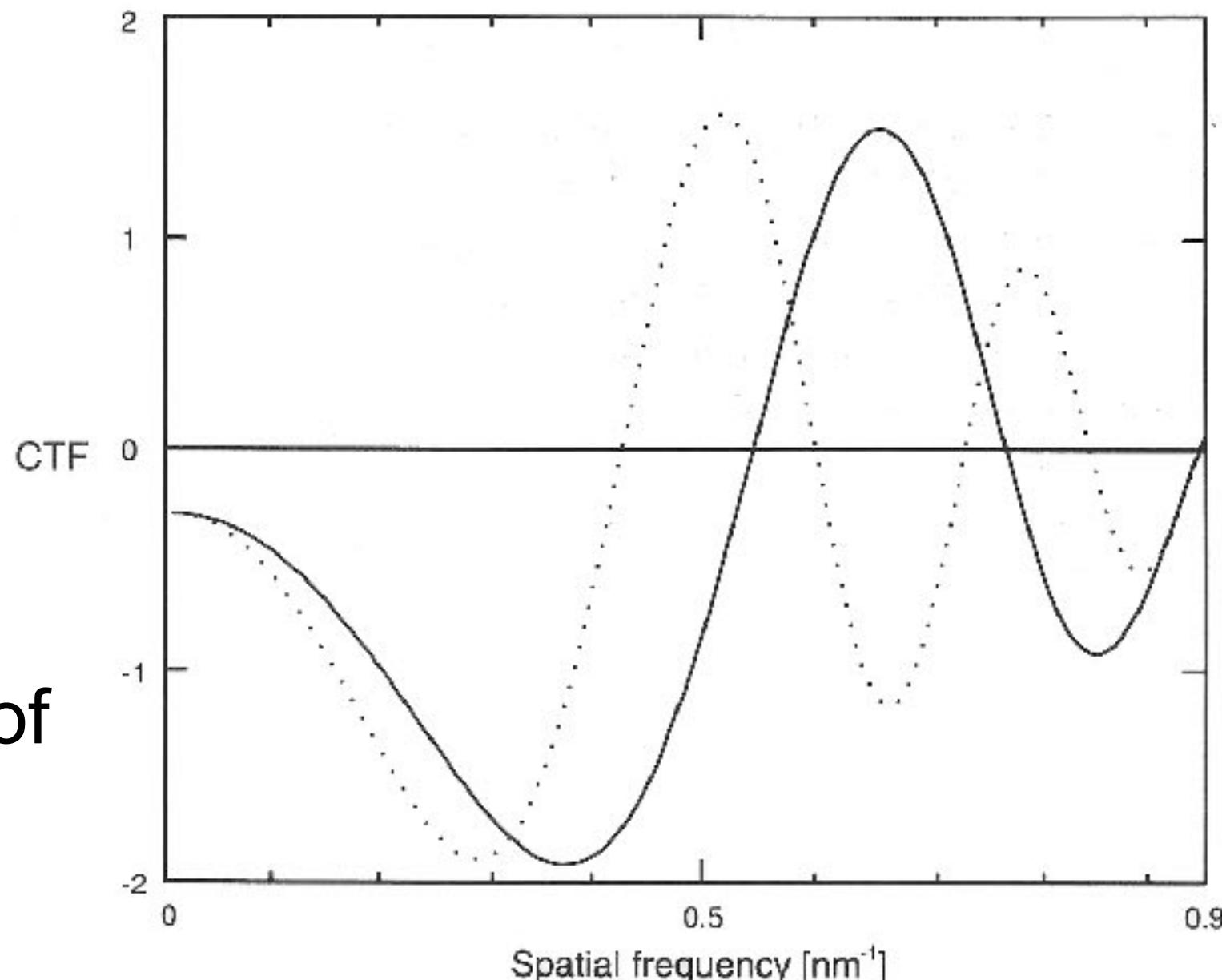
# Contrast Transfer Function (CTF)

$$CTF = \sin(2\pi\chi k)$$

The intensity of a recorded image is directly related to the projection of specimen (good!) but modified by the FT of CTF (bad!).

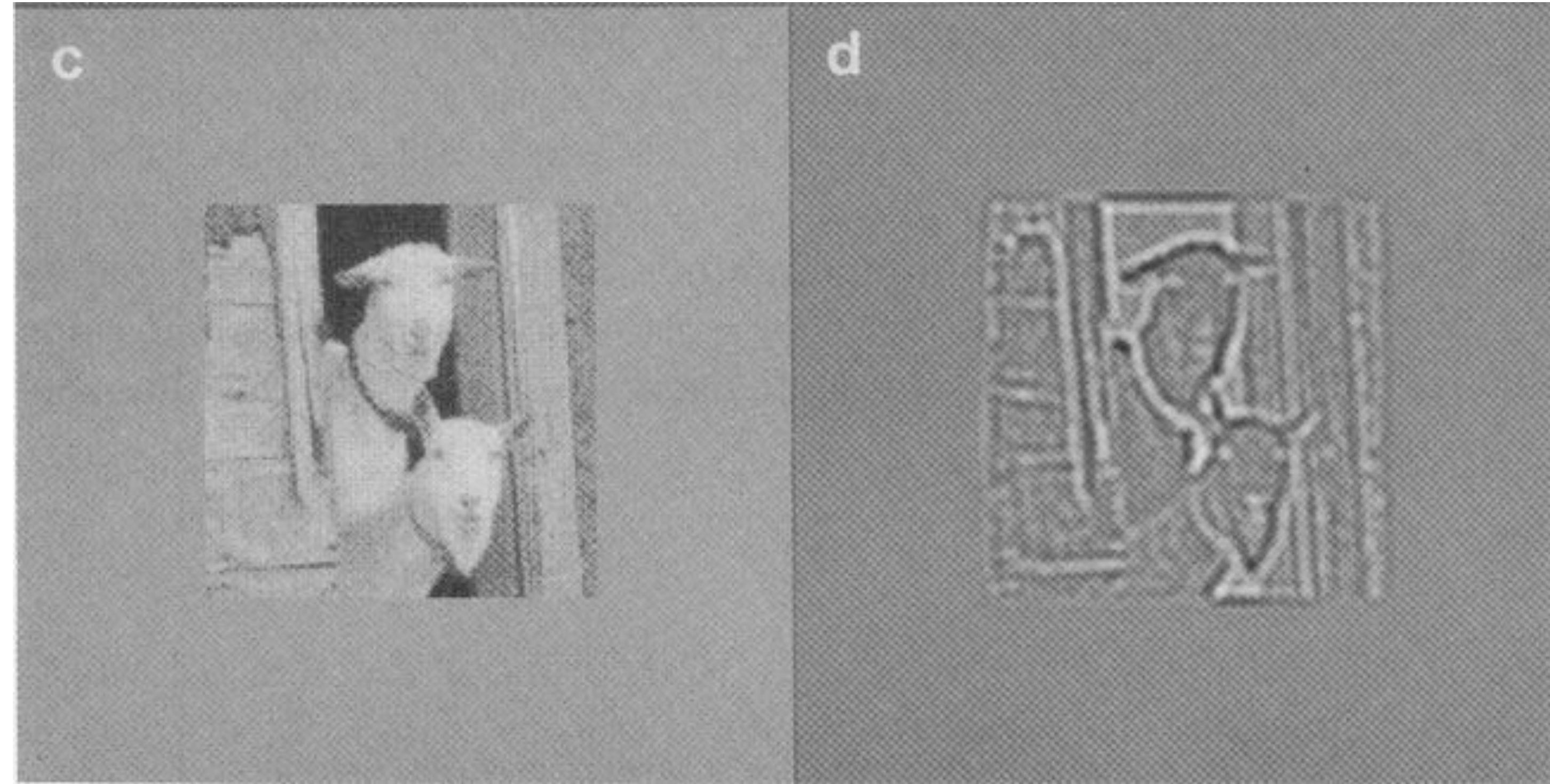
$$I_i(\vec{r}) = \psi_i(\vec{r})\psi_i^*$$

$$= 1 + 2\Phi(-\vec{r}) \otimes J_0(\vec{r}) \otimes F^{-1}(CTF) \quad (12)$$



WebCTF: <http://jiang.bio.purdue.edu/ctfsimu>

# What is this CTF thing anyway and why do I care?

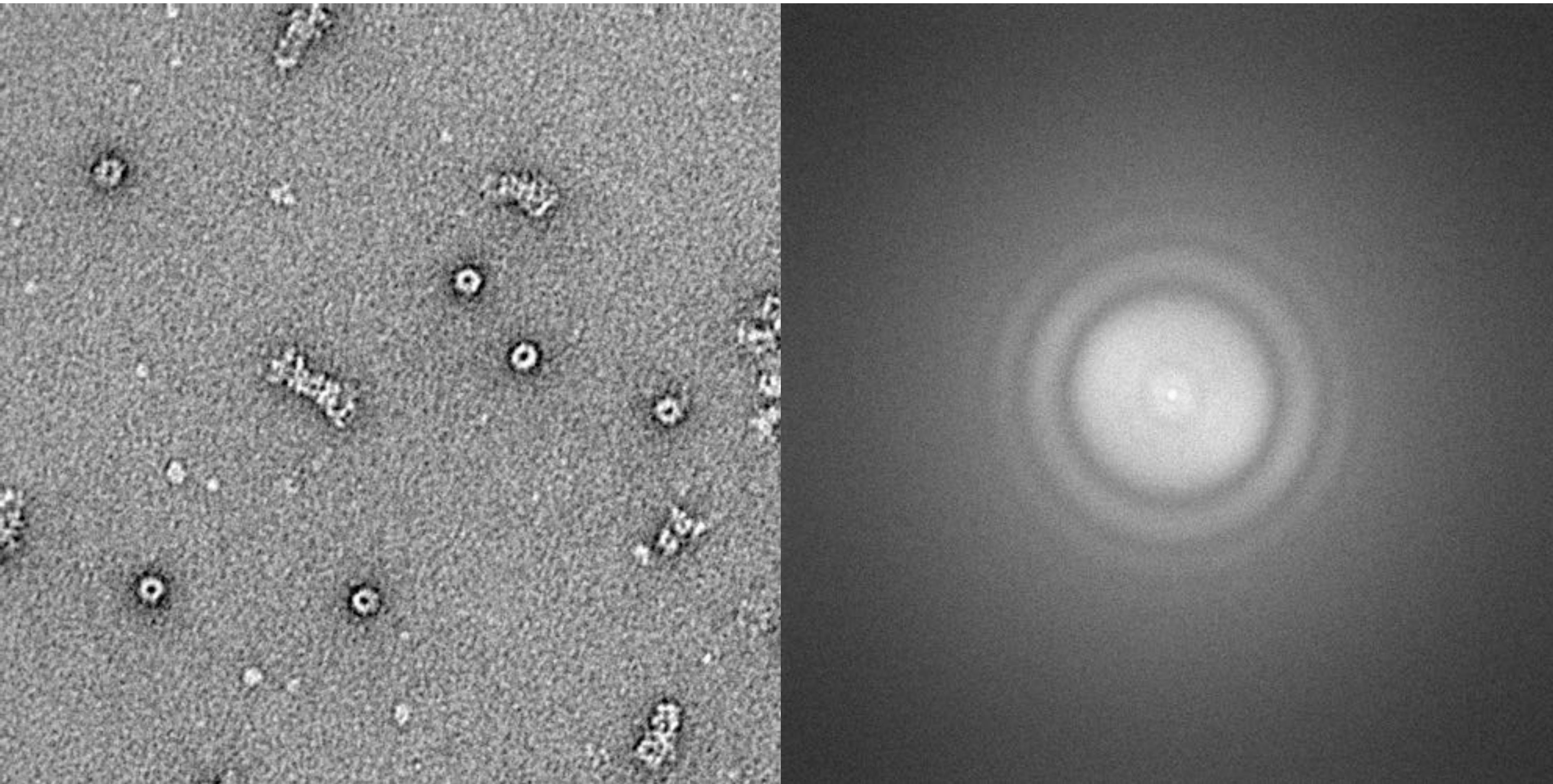


Distortions of CTF to the image are:

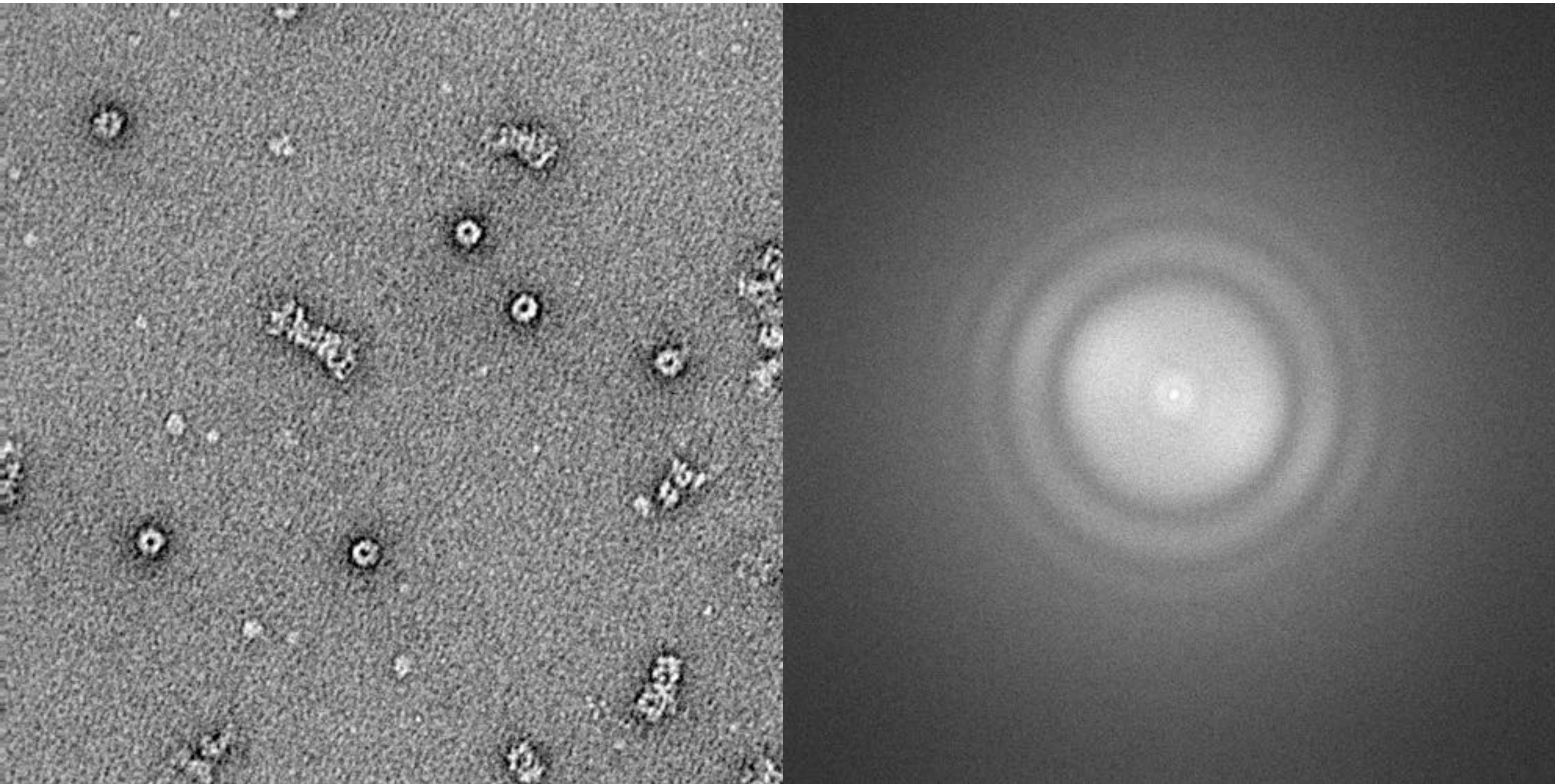
- 1) Contrast reverve of large area; 2) diminished contrast in large area; 3) edge enhancement and 4) appearance of fringes along the borders.

From Joachim Frank

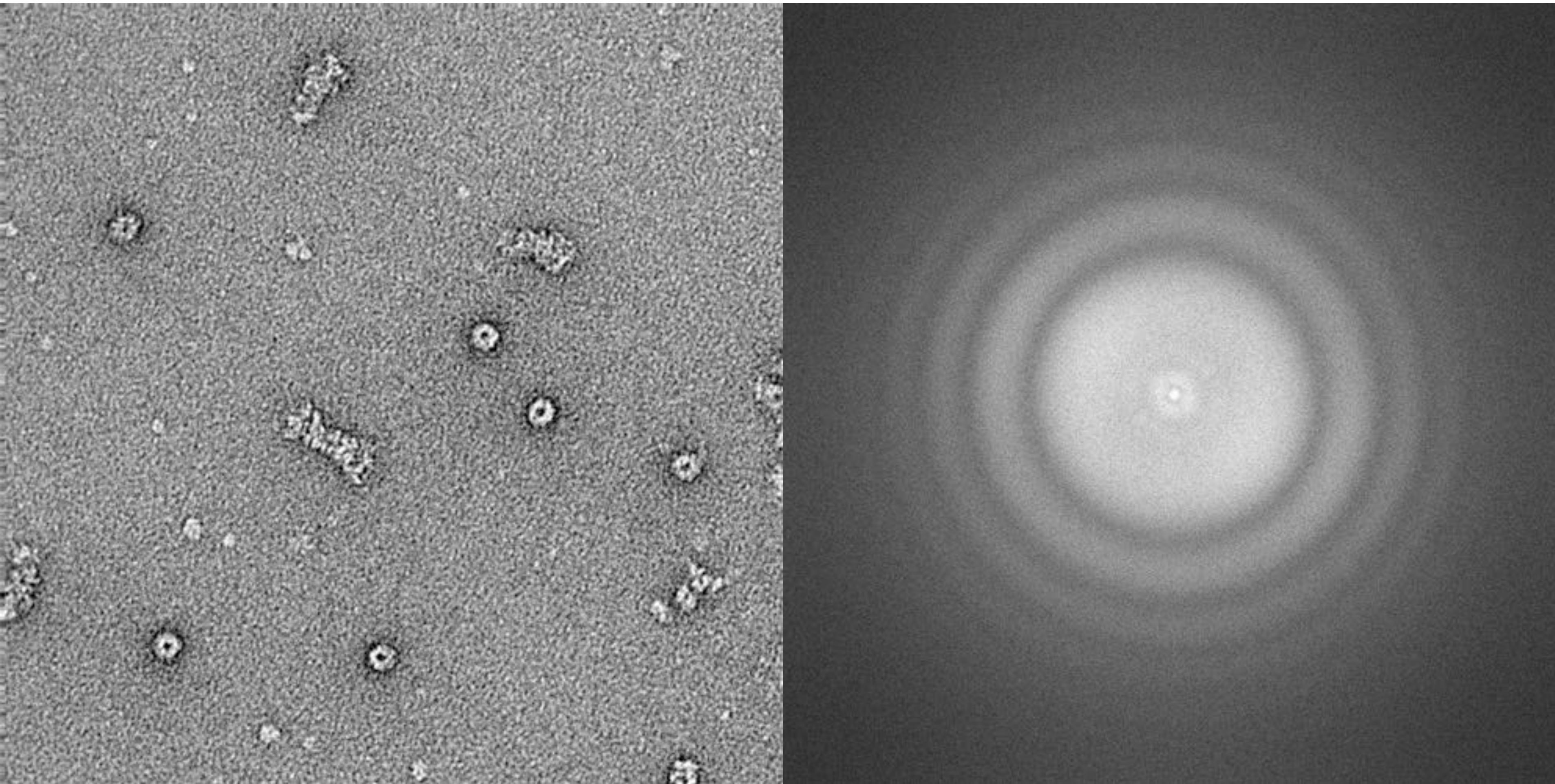
**Defocus -2 $\mu$ m**



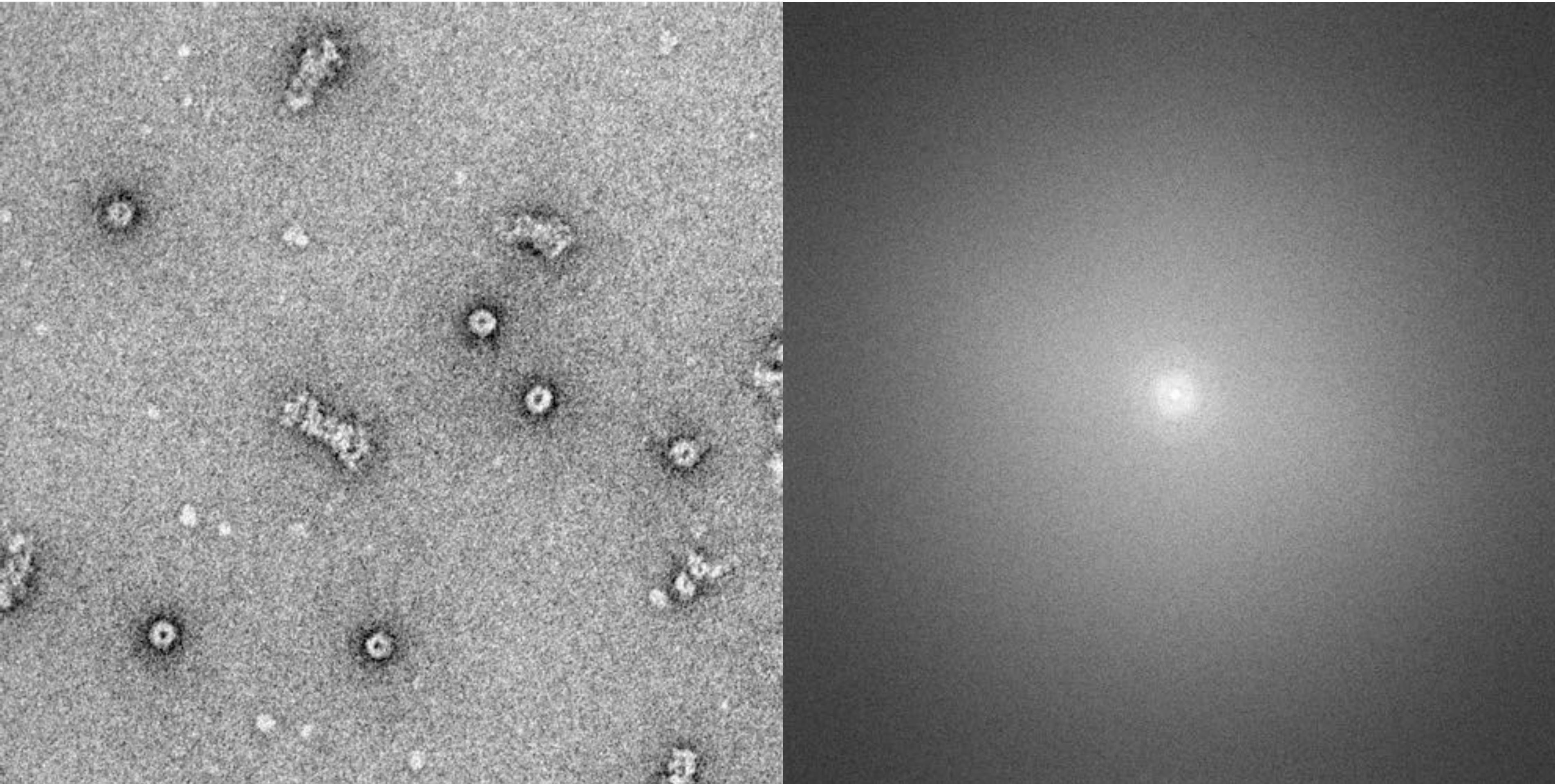
**Defocus -1.5 $\mu$ m**



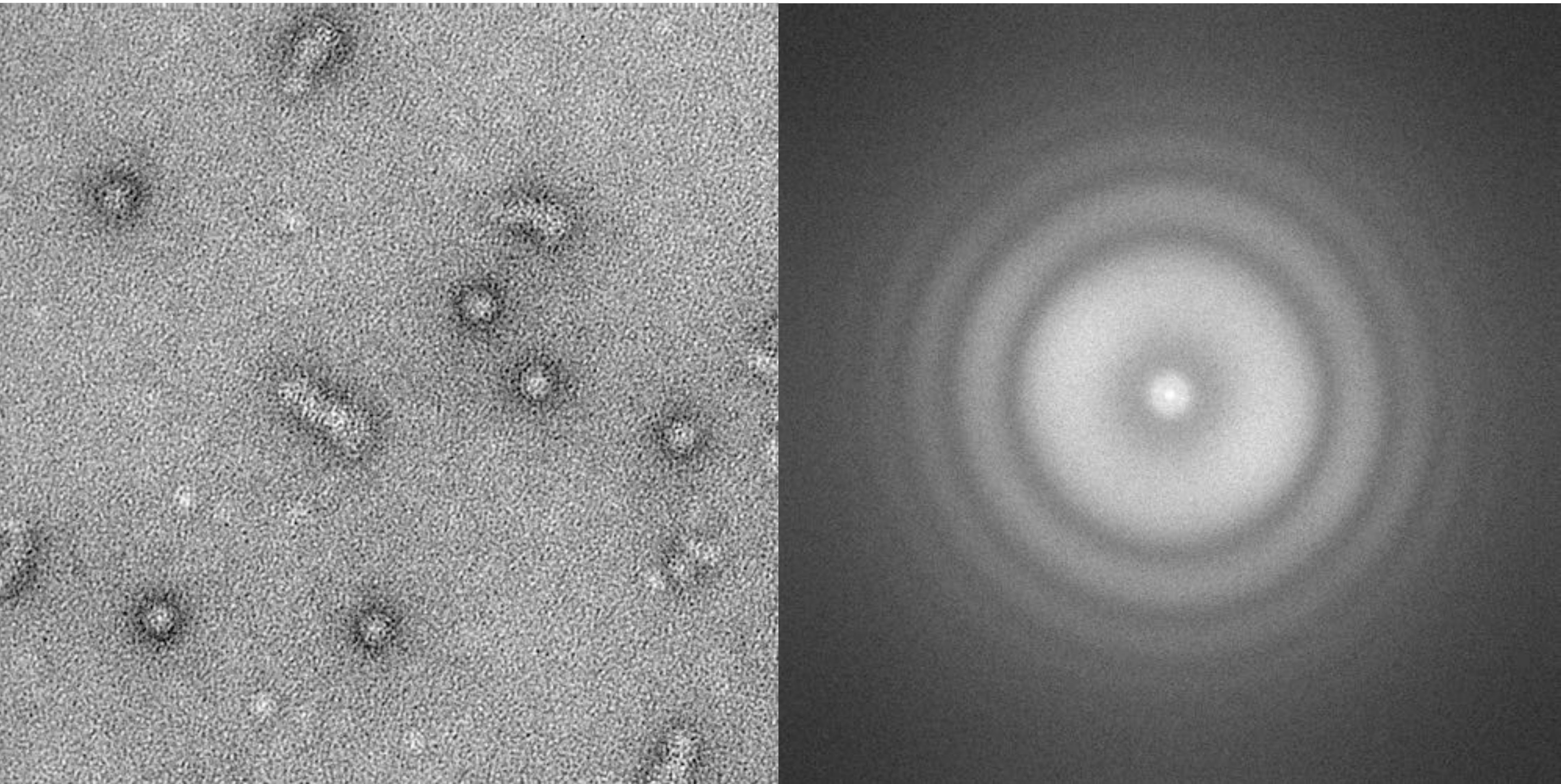
**Defocus -1 $\mu$ m**



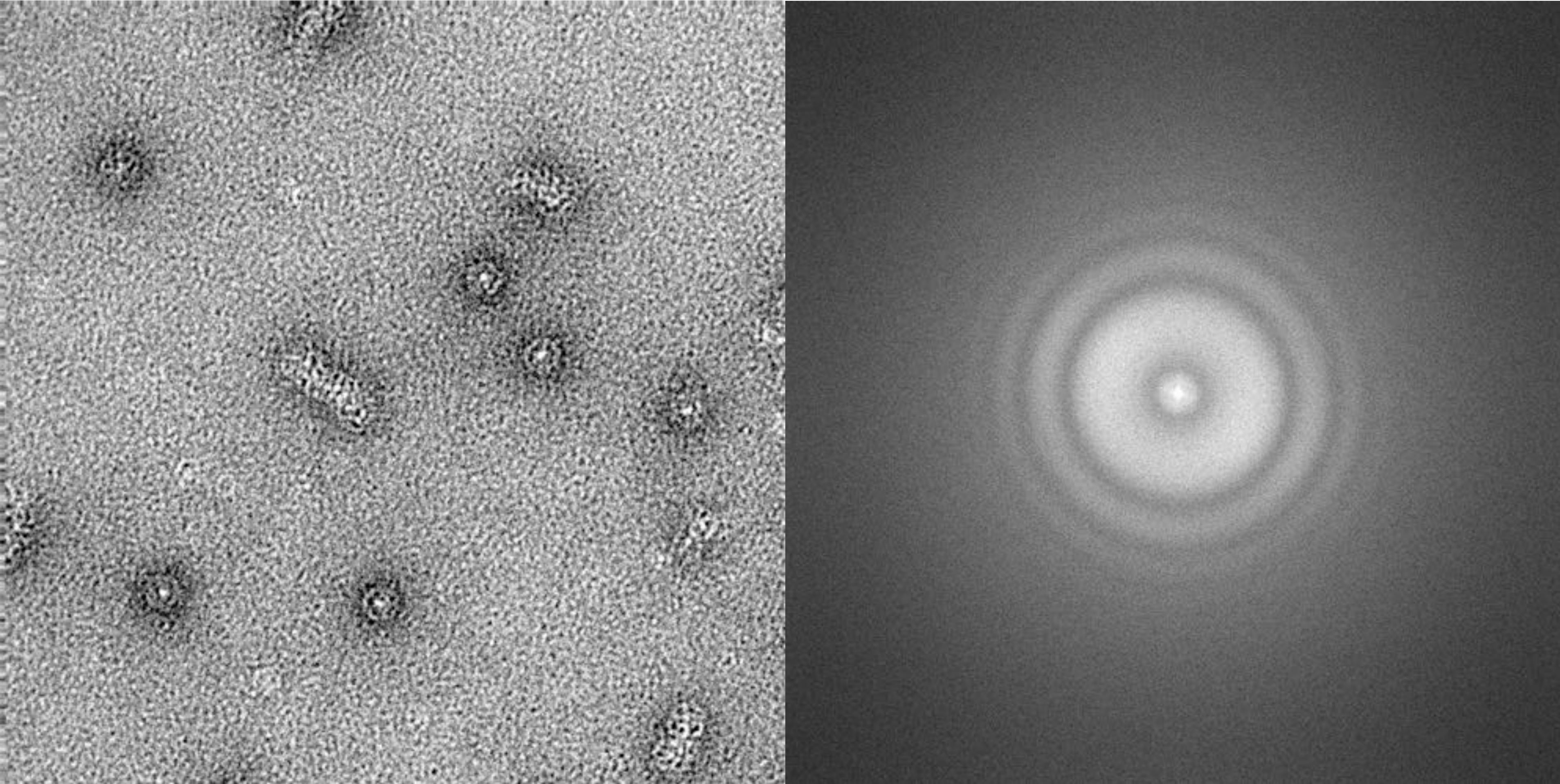
**Defocus ~0 $\mu$ m**



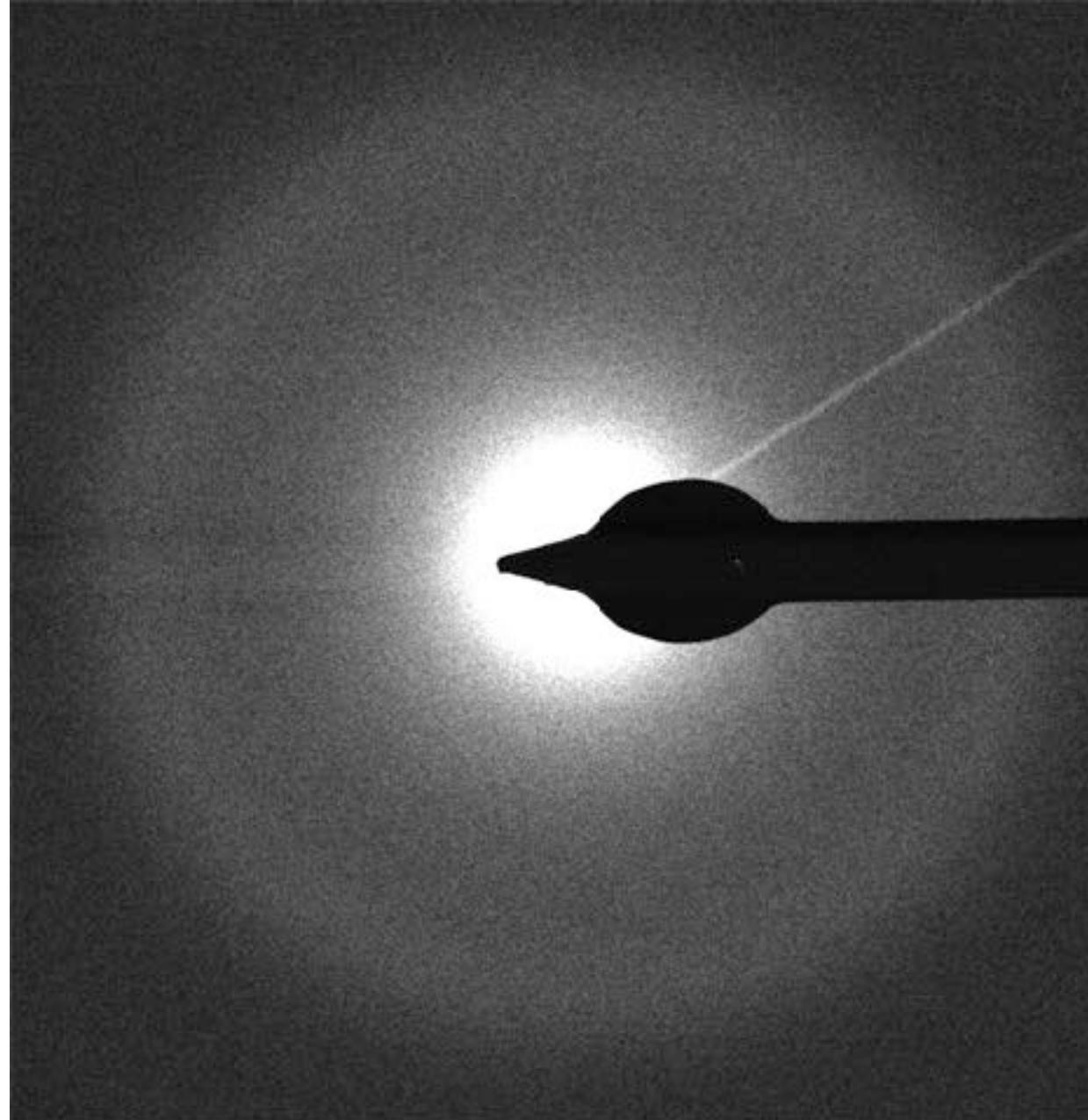
**Defocus +1 $\mu$ m**



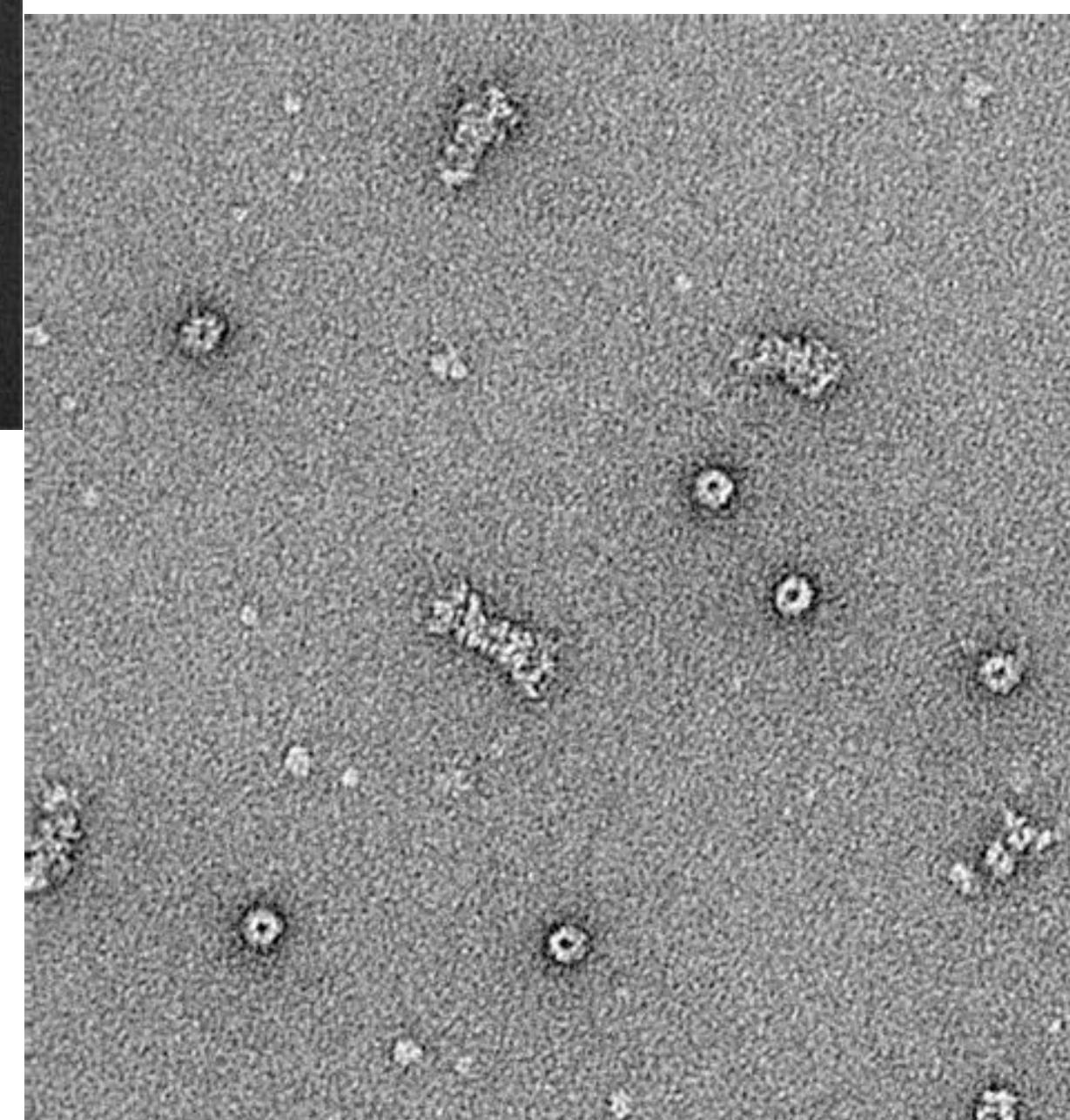
**Defocus +2 $\mu$ m**



# Diffraction, image and power spectra

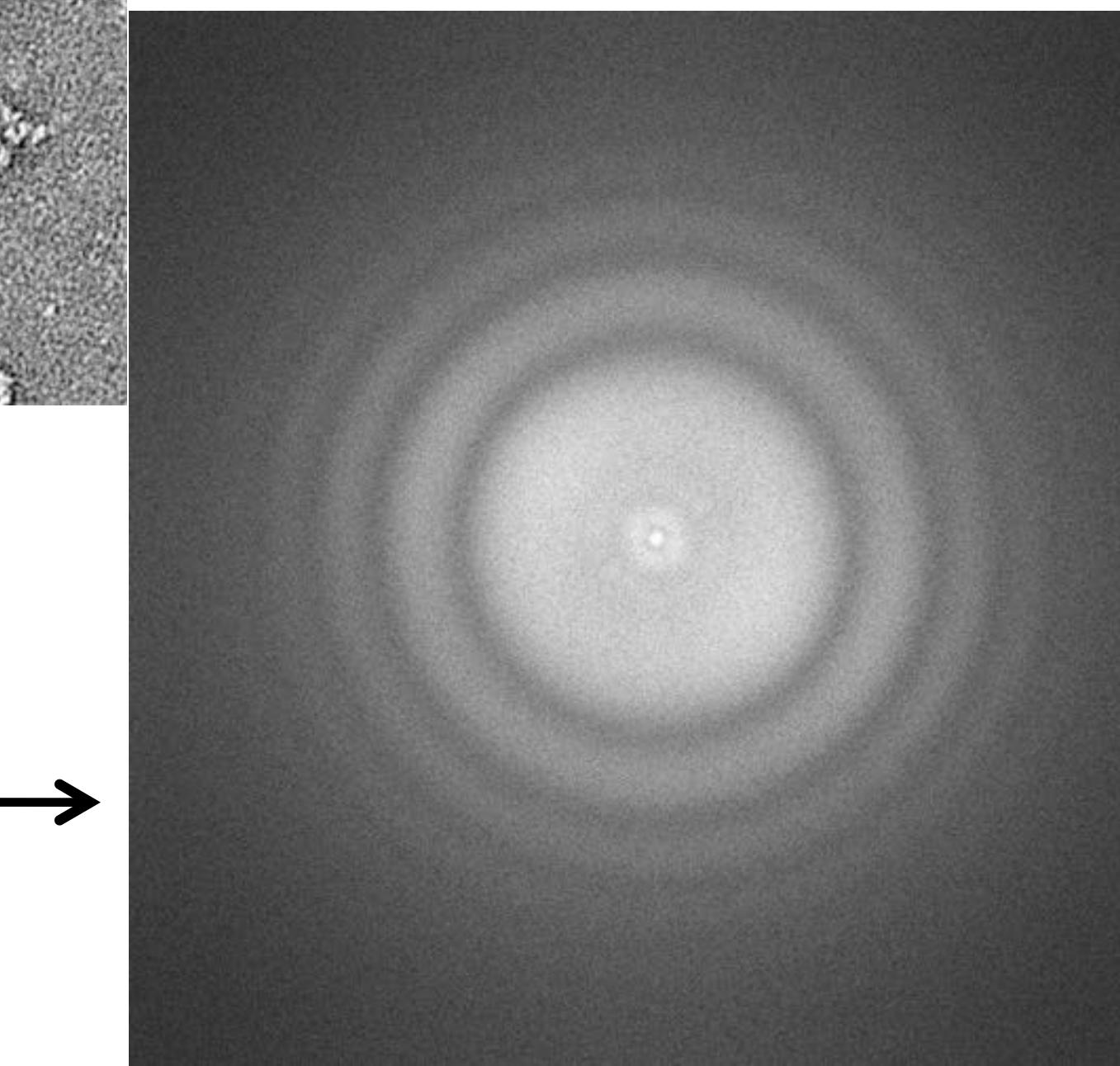
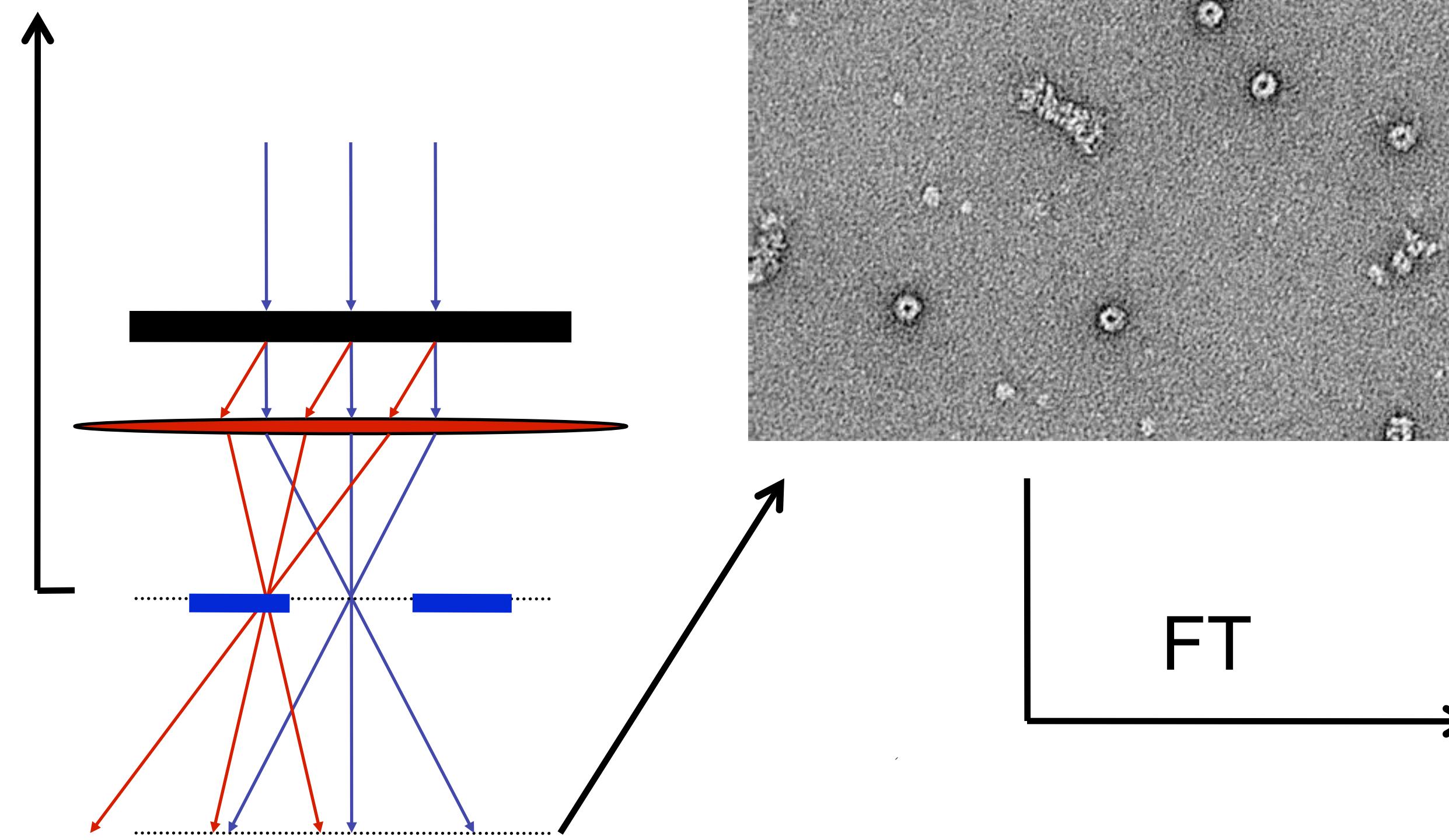


diffraction



image

Fourier power  
spectrum



## Determine CTF

Model

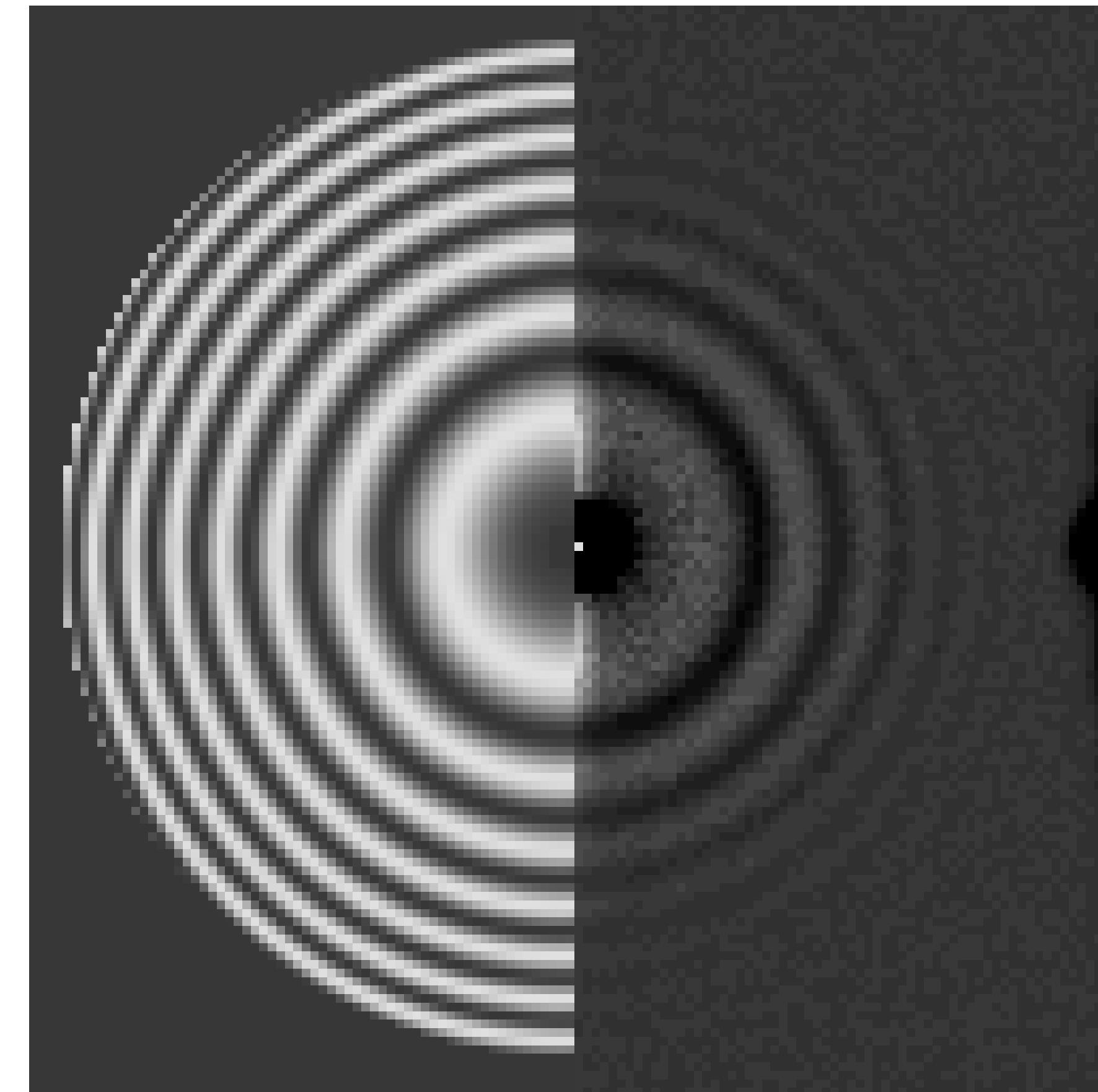


Image power  
spectrum

Experiment

$$E = 120 \text{ kV}, \Delta f = 21000 \text{ \AA}, C_s = 2 \text{ mm}, A = 0.15$$

# Atomic resolution imaging with TEM

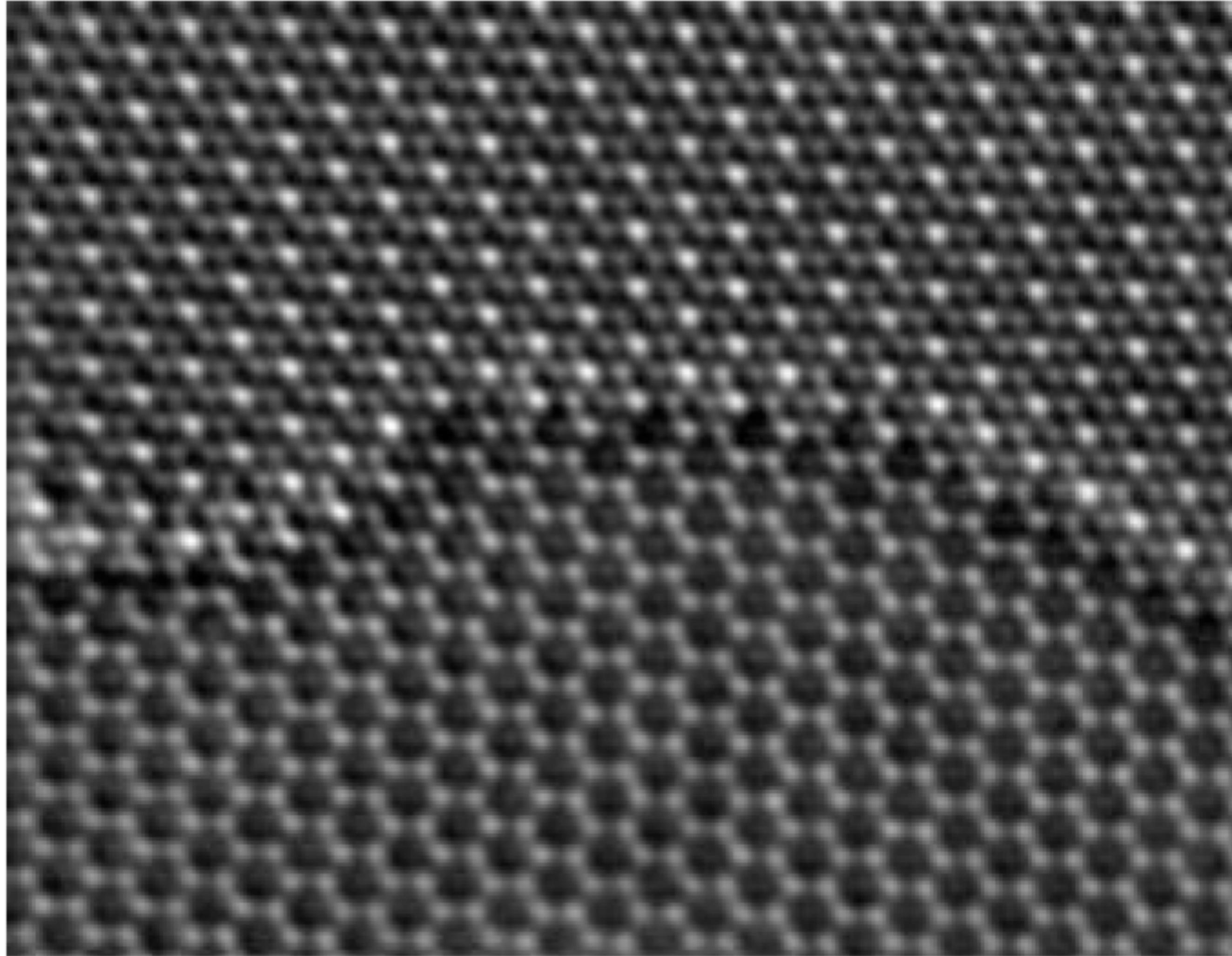
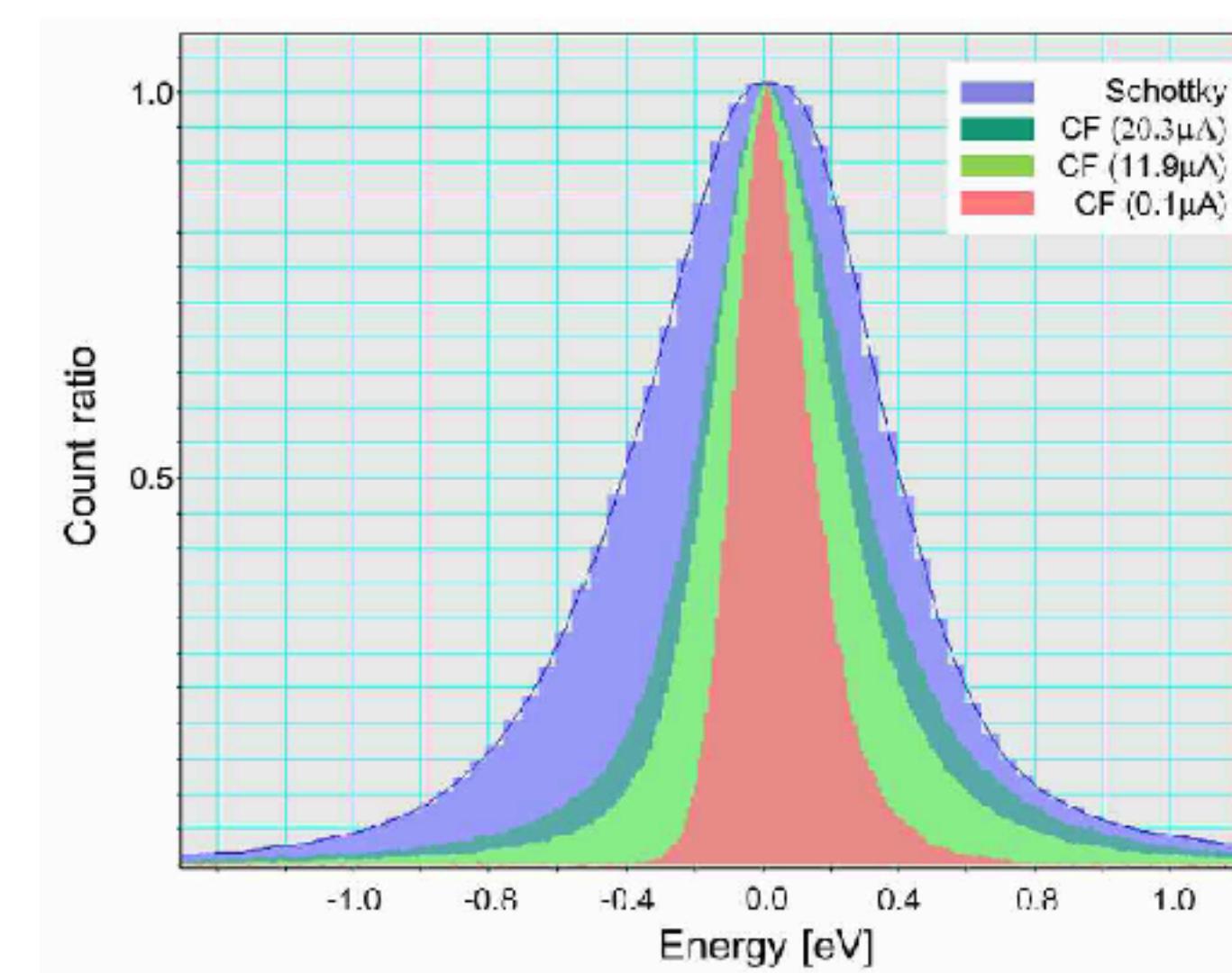
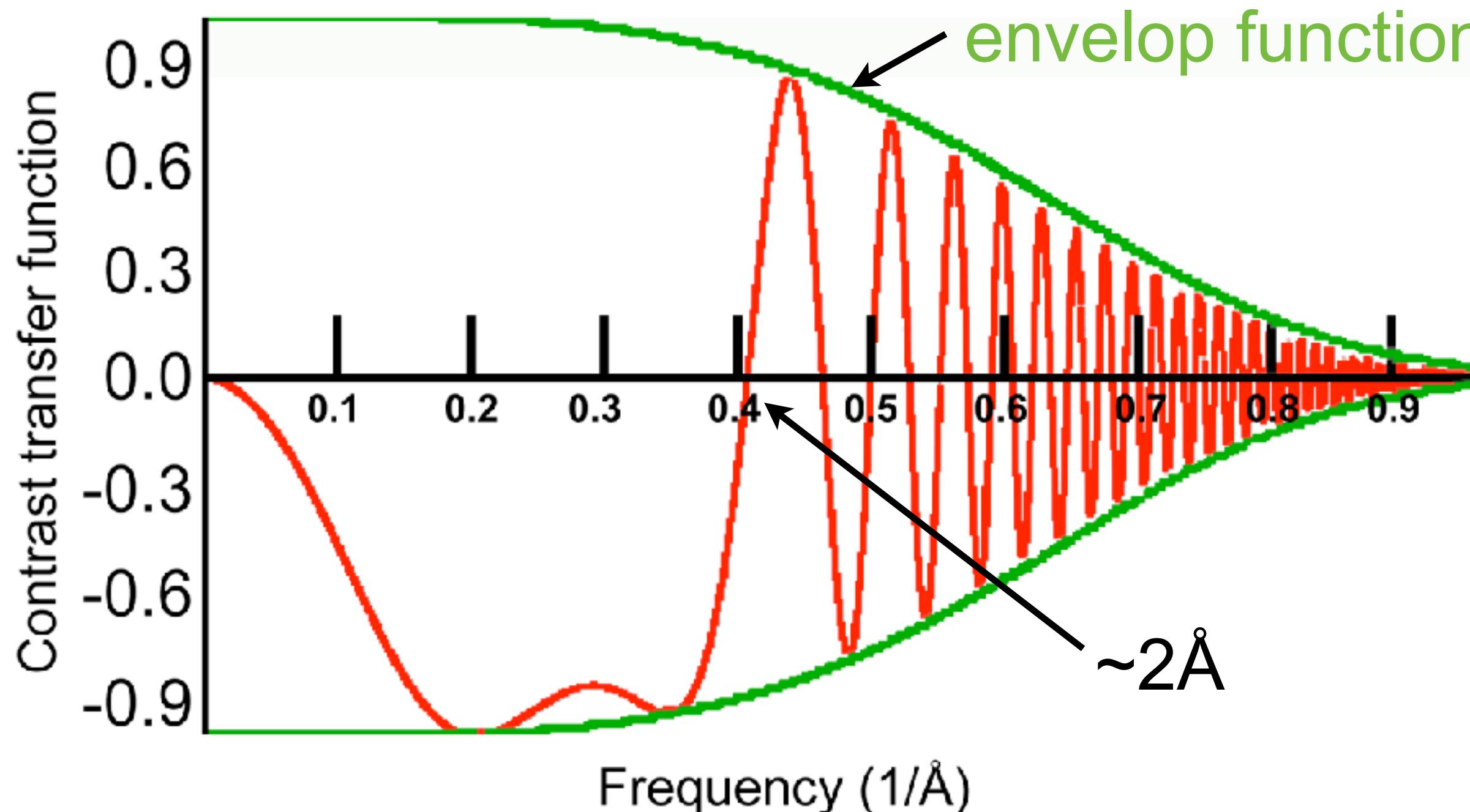


Image of graphene, Nature Mat, 2011, **10**, 165

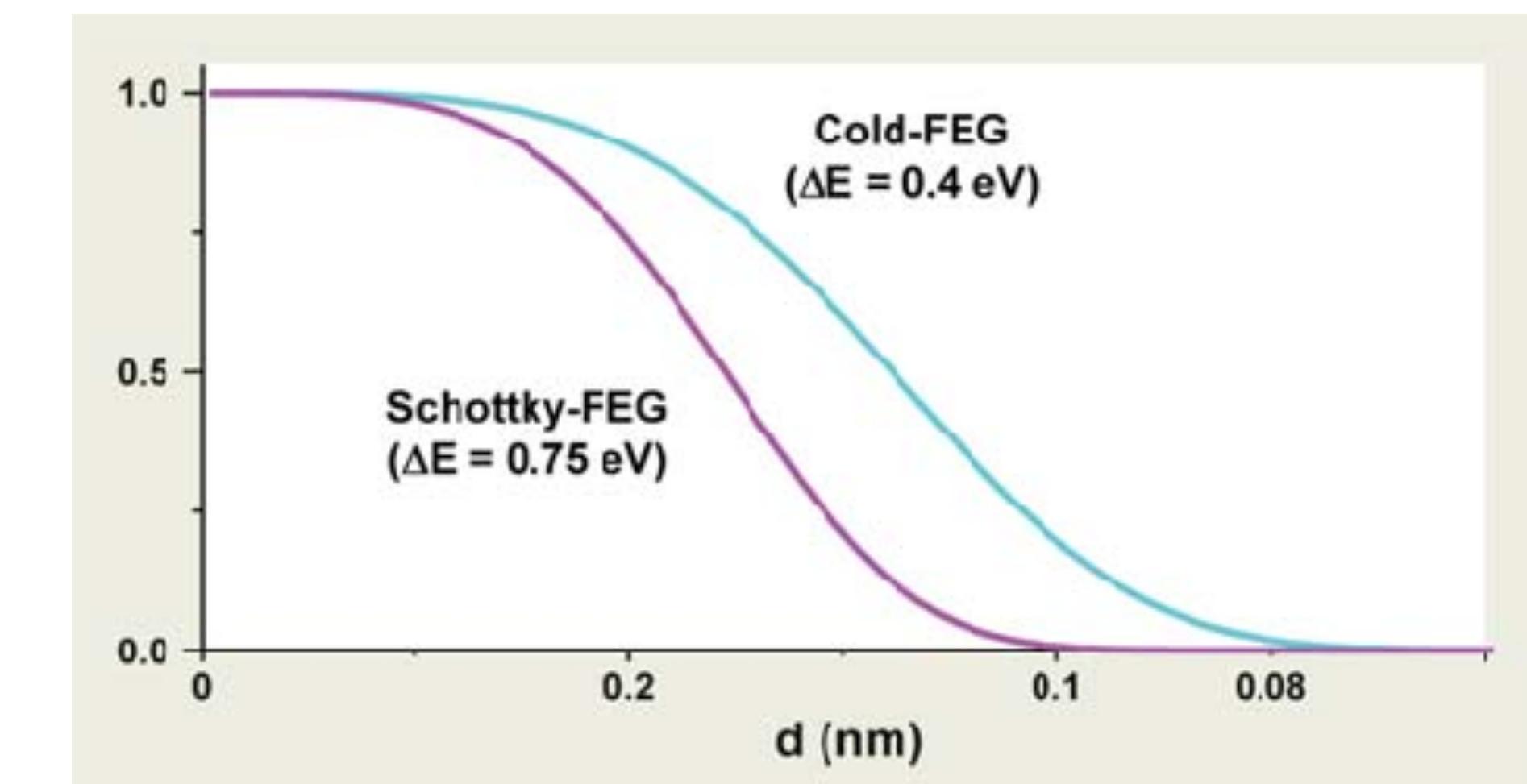
Electron optic system of a modern electron microscope is of sufficient quality to image radiation resistant material (typically inorganic) at atomic resolution ( $\sim 2\text{\AA}$  or better).

# Determinants of resolution

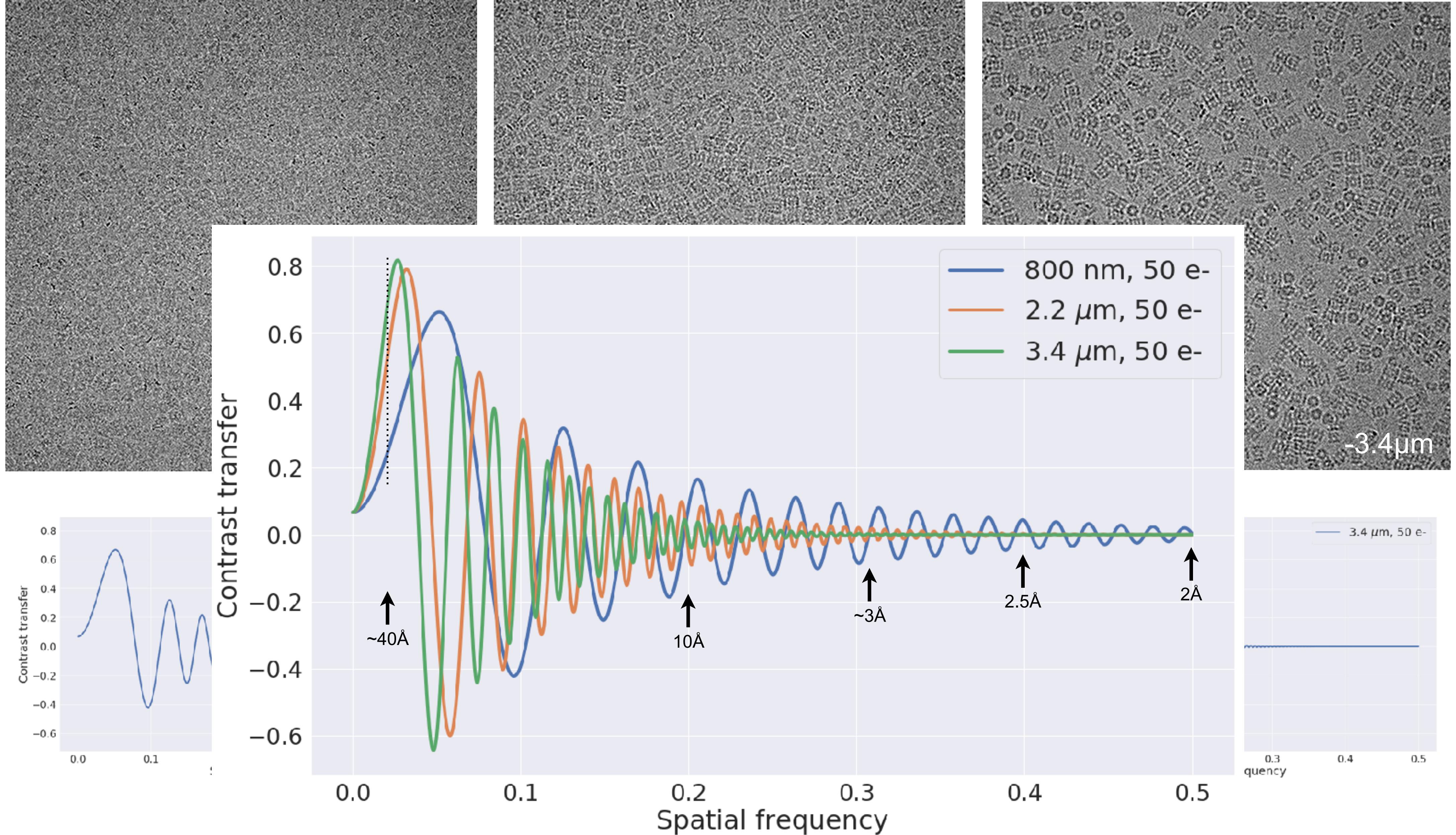


- Envelop function determines the information limit of a micrograph;
- Envelop function itself is shaped by defocus, beam spacial coherence,

$$D(\vec{k}) = e^{-\frac{1}{2}\pi^2 \Delta^2 \lambda^2(\vec{k})^4} e^{-\pi^2 \alpha^2(\vec{k})^2} \left[ \varepsilon + C_s \lambda^2(\vec{k})^2 \right]^2$$



# Influence of CTF on image



# Influence of sample thickness

