

Journal

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Jul. 16, 2024 — Tuesday

About Voronoi tessellation and characterization techniques in softmatter. In the article Yang, Zou, and Yu (2002)

They analyse a system with 5000 particles and vary the radius from 100 to 0.1m and analyse the number of edges for each polyhedron face, number of faces for each polyhedron face, perimeter, area of a polyhedron face and perimeter, area and volume of a polyhedron. In a nutshell, they said that the topological and metric properties of Voronoi polyhedra are quantified as a function of particles size and packing density. Edges, vertices, volume, area in terms of the particle size and packing fraction. According to the results of the article, the average face number decreases when the particle size and packing density decreases. The distributions of face number and edges become broader and more asymmetric. On the other hand, the average perimeter and area of polyhedra increases and the distributions of polyhedron surface area and volume become more flat and can be described by the log-normal distribution. Finally, the average sphericity coefficient of Voronoi Polyhedra varies with packing density. [The sphericity, Ψ , of an object is the ratio of the surface area of a sphere with the same volume to the object's surface area $\Psi = \pi^{1/3} (6V_p)^{2/3} / A_p$].

They mention the following laws: Aboav-Weaire's law and Lewis's law.

In the article Lazar, Lu, and Rycroft 2022 in the section of Stability they state that: When geometric quantities are used for classification purposes, thresholds for the range of possible values must be chosen, on the other hand, topological ones can change abruptly under small perturbations of particle coordinates.

How we can relate the topological and metric properties with reholological properties?

Relation between the interatomic interaction with the geometric properties. I think that there is a relation between the superficial area of the voronoi faces with pressure between two particles.

Thinking about the Lennard-Jones potential in three dimensions, it creates an sphere of potential. It can be computed the pressure in a plane that intersect that sphere.

I think that, that relation, between the potential and the intersepte plane, will help to create a conecction between rehological properties and structural stuff.

$$\vec{P}_{VC} = -\frac{d}{dr} L J(r) |_{r \in R^2} \frac{1}{A_{VC}} \hat{e}_r,$$

where A_{VC} is the area of the shared plane between two voronoi cells and the force is evaluated at that plane. The VC indicates, "Voronoi Cell".

Now, this has the same units of the stress tensor.

To introduce the direction of the plane into the pressure, we take dot product with the spherical coordinate system, and then multiply by the normal vector of the plane,

$$\vec{P}_{plane} = [\vec{P}_{VC} \cdot (\hat{e}_r, \hat{e}_\phi, \hat{e}_\theta)] \hat{n}_{plane}$$

Create an histogram of the normal vector. Histogram of the dot product between the normal vector with the cartesian reference.

The better way to analyse the macro characteristics is an histogram.

About packing density: Using the volumen of voronoi polyhedra. It can be said that a system is jammed, when the mean of the volumens of the voronoi polyhedra tends to a minimum.

About the simulations. 50 simulations with 500 patchy particles took 24 hrs. This is 500 Monomers and Crosslinker. All of them has 50 CL and 450 MO, hence, 200 patches for the 50 CL and 900 patches for MO, hence, 1600 particles in total.

I have no idea how to improve the time of the simulation. Each simulation takes between 27-28 mins.

★ Jul. 17, 2024 — Wednesday

Key point of the reunion:

- Start the script for shear deformation
- Keep in mind to vary the following parameters:
 1. Cross-Linker concentration
 2. Size of the box (Packing density)
 3. The energy of the Patch-Patch interaction.
- Analyze the density distribution of the voronoi parameters
- Make an animation, where the color of the particle represent the volume of the voronoi cell and share the script.
- Hint: Directores

About the idea of the normal vector: They do not see a great future.

The deformation script is done, basically I copy the code from Felipe.

Jul. 19, 2024 — Friday

Yesterday I finish the scripts for the shear deformation and investigate some papers that help me to understand what I'm trying to do for the thesis project.

I found the article by Sheiko and Dobrynin 2019 and here are some "main" points:

- Materials response to deformation is expressed by a stress-strain curve measured at controlled strain rate and temperature.
- A set of stress-strain curves constitute the "mechanical phenotype".
- The dependency of those curves help us to understand the molecular relaxation processes, such as conformational transformation, bond scission and molecule displacement.
- In the article, they classify a material, based on their stress-strain curves, in three categories: Thermoplastics, Biological networks and Synthetic network.
- Feature of interest for the article: Firmness is the stiffness enhancement in response to deformation and is one of nature's defense mechanisms preventing accidental organ rupture.
- Unlike thermoplastics, the mechanical properties are predominantly controlled by molecular interactions.
- Elastomers, the mechanical properties are controlled by the changes in the system configuration, by unravelling network strands at nearly constant mass density.
- A network architecture is defined by: Connectivity, dimensions and flexibility of network strands.
- The weak dependence on chemical compositions helps to manipulate the mechanical response.
- In the article they focus on the Shear modulus.
- Topological defects: Chain entanglements, dangling chain ends, loops, multiple strands and side chains.
- Search of definitions:
 1. Shear modulus:
 2. Cross-link functionality:
 3. Degree of polymerization: Number of monomeric units (MO and CL)
 4. Kuhn length: A theoretical treatment, developed by Werner Kuhn, in which a real polymer chain is considered as a collection of N Kuhn segments each with a Kuhn length b . Each Kuhn segment can be thought of as if they are freely jointed with each other.
- The softness at small deformations is manipulated by changing the cross-link density.
- Linear chain Networks
 - The cross-link density is THE control parameter

- The Young's modulus can be tune by introducing loops and dangles.
- Modify the junctions can controll the network topology
- The cross-link functionality can be increased by adding dendrimes, colloidal particles and microphase domains.
- Their equilibrium characteristics are mostly determined by the degree of polymerization of the chemical network strand.
- Strand entanglements can create a loweer limit for the shear modulus and upper limit for the elongation-at-break.
- Softness can be achieved by solvent swelling.
- Swelling leads to strand extensions and therefore firmness enhacement.
- Dual Networks
 - Networks with permanent cross-links and dynamic cross-links.
 - Permamnent cross-link are responsible for elastic properties and sample shape
 - Dynamic cross-link regulate energy dissipation and strain-rate response.
 - Dynamic networks depen on strain rate, permanent network does not depend on strain rate.
 - At higher rates, physical cross-links do not have enough time to break and behave as permanent cross-links, resulting in high modulus and strength.
 - **To characterize the dynamic mechanical properties of the dual networks at small deformations, evolution of the stress relaxation Young's modulus $E(t)$ during uniaxial extension at a constant temperature is calculated by using a linear viscoelasticity approximation.**
- Network with Brush-Like strands: I didn't read it with detail, beacuse it is out of my scope/comprehension/interest at the moment. It will be great to read it with more knowledge in the area.

Need to check Dealy and Larson 2006 for more information about polymer structures and their characterization.

Now, in the sense, that the curve of strain-stress is essential to characterize the material, I search in lammps how to compute the stres and strin in the simulation. What I found is an page where there explain how to Calculate elastic constants and according to that page, the elastic constants (strain and stress) are related as follows:

$$S_{ij} = C_{ijkl} e_{kl},$$

where the repeates indices impley summation. S_{ij} are the elemnts of the symmetric stress tensor, e_{kl} are the elements of the symmetric strain tensor and C_{ijkl} are the elemnts of the fourth rank tensor of elastic constants.

Then, they state that the elastic constant can be computed at *zero temperature* or *finite temperature* and metiond the following methods for doing it in LAMMPS,

- At zero temperature

1. The C_{ijkl} are estimated by deforming the simulation box in one of the six directions using `change_box` command. Then compute the changes in the stress tensor.

- At finite temperature

1. Exploit the relation between elastic constants, stress fluctuations, and the Born matrix. Ray and Rahman 1984
2. Measure the change in average stress tensor in an NVT simulations when the cell volume undergoes a finite deformation.
3. Sample the triclinic cell fluctuations that occur in an NPT simulation.

Finally, in the article Clavier et al. 2017 a nice review of the advantages and disadvantages of all of these methods is provided.

So the next week I will check those articles and the examples scripts in lammps.

Possibly, during the weekend's I transfer old entries of multiple daily log in Obsidian, notebooks, Git and Discord.

References

- Clavier, Germain et al. (June 2017). "Computation of Elastic Constants of Solids Using Molecular Simulation: Comparison of Constant Volume and Constant Pressure Ensemble Methods". *Molecular Simulation* **43**.17, pp. 1413–1422. DOI: [10.1080/08927022.2017.1313418](https://doi.org/10.1080/08927022.2017.1313418). URL: <https://ifp.hal.science/hal-01702896> (visited on 07/20/2024).
- Dealy, J.M. and R.G. Larson (Jan. 1, 2006). "Structure and Rheology of Molten Polymers—From Structure to Flow Behavior and Back Again". *Structure and Rheology of Molten Polymers: From Structure to Flow Behavior and Back Again*, pp. 1–516. DOI: [10.3139/9783446412811](https://doi.org/10.3139/9783446412811).
- Lazar, Emanuel A., Jiayin Lu, and Chris H. Rycroft (June 1, 2022). "Voronoi Cell Analysis: The Shapes of Particle Systems". *American Journal of Physics* **90**.6, pp. 469–480. ISSN: 0002-9505. DOI: [10.1119/5.0087591](https://doi.org/10.1119/5.0087591). URL: <https://doi.org/10.1119/5.0087591> (visited on 06/26/2024).
- Ray, John R. and A. Rahman (May 1, 1984). "Statistical Ensembles and Molecular Dynamics Studies of Anisotropic Solids". *The Journal of Chemical Physics* **80**.9, pp. 4423–4428. ISSN: 0021-9606. DOI: [10.1063/1.447221](https://doi.org/10.1063/1.447221). URL: <https://doi.org/10.1063/1.447221> (visited on 07/20/2024).
- Sheiko, Sergei S. and Andrey V. Dobrynin (Oct. 22, 2019). "Architectural Code for Rubber Elasticity: From Supersoft to Superfirm Materials". *Macromolecules* **52**.20, pp. 7531–7546. ISSN: 0024-9297. DOI: [10.1021/acs.macromol.9b01127](https://doi.org/10.1021/acs.macromol.9b01127). URL: <https://doi.org/10.1021/acs.macromol.9b01127> (visited on 07/18/2024).
- Yang, R. Y., R. P. Zou, and A. B. Yu (Apr. 3, 2002). "Voronoi Tessellation of the Packing of Fine Uniform Spheres". *Physical Review E* **65**.4, p. 041302. DOI: [10.1103/PhysRevE.65.041302](https://link.aps.org/doi/10.1103/PhysRevE.65.041302). URL: <https://link.aps.org/doi/10.1103/PhysRevE.65.041302> (visited on 06/27/2024).