

# Asymptotic properties of carbon based composite materials used in latent heat thermal energy storage systems

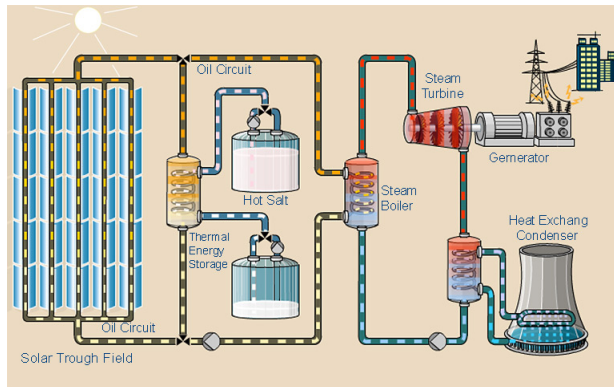
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LMAC, Université de technologie de Compiègne  
16 Décembre, 2015

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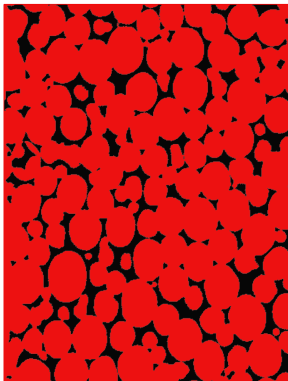
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# Latent heat thermal energy storage systems

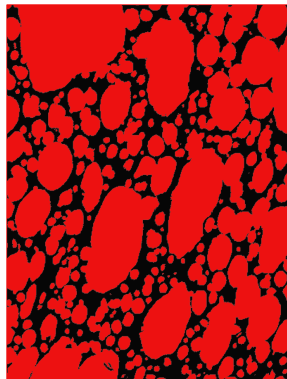


# The studied composite materials

KL1\_250



KD1



These composites are introduced by V. Canseco, Y. Anguy, J. J. Roa and E. Palomo in [1].

# Thermo-physical properties of the composites

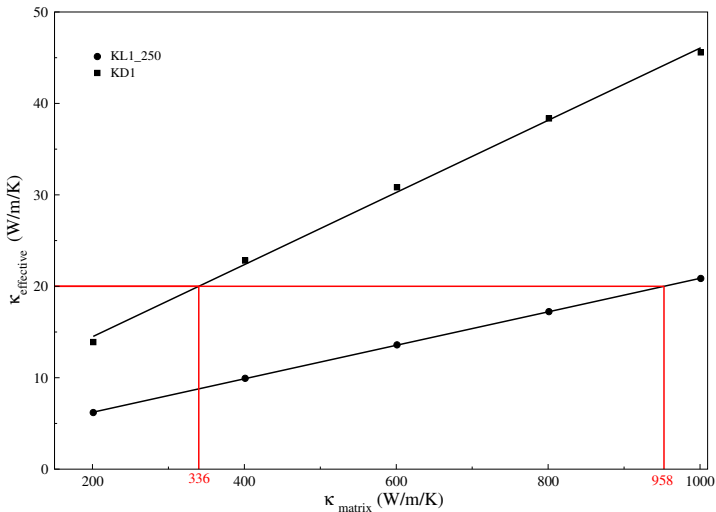
Parameter	KL1_250	KD1
porosity(%)	82	70
$\kappa_{PCM}$ (W/m/K)	1	1
$C_{p_{PCM}}$ (J/kg/K)	4544	4544
$C_{p_{Foam}}$ (J/kg/K)	1402	1402
$\rho_{PCM}$ (J/kg/K)	1634	1634
$\rho_{Foam}$ (J/kg/K)	1318	1318
$\Delta h_m$ (KJ/kg)	480	480

the porosity of a composite is defined as  $\mathcal{E} = \frac{V_{pores}}{V_{Total}}$

We want  $\kappa_{eff}$  the same for both composites

We know that  $\kappa_{eff} = a * \kappa_{Foam} + b$

# The effective thermal conductivity of the composites



# Thermal energy storage capacity

For 2 different 2D rectangular samples, of dimensions:  $x_1 \times y_1$  and  $x_2 \times y_2$  with porosities  $\mathcal{E}_1$  and  $\mathcal{E}_2$  with the same latent heat of fusion  $\Delta h_m$ , their thermal energy storage capacities are

$$Q_1 = \mathcal{E}_1 x_1 y_1 \Delta h_m$$

and

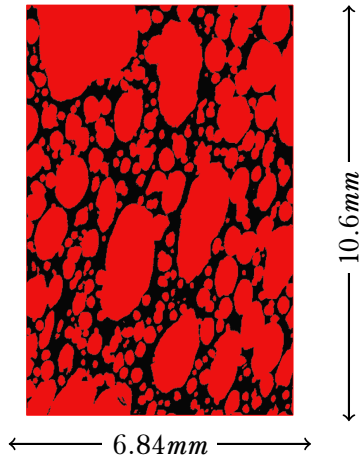
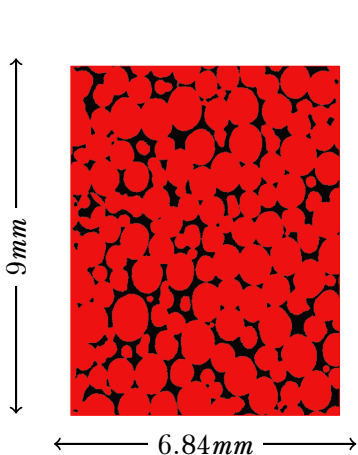
$$Q_2 = \mathcal{E}_2 x_2 y_2 \Delta h_m$$

hence to have  $Q_1 = Q_2$  with  $x_1 = x_2$  we need to have  $\mathcal{E}_1 y_1 = \mathcal{E}_2 y_2$

In our case  $\mathcal{E}_{KL1\_250} = 82\% > \mathcal{E}_{KD1} = 70\%$

we need to take  $y_{KD1} > y_{KL1\_250}$

## Dimensions of the samples





# The model

$$\left\{ \begin{array}{l} \partial_t H(T) - \operatorname{div}(\kappa \nabla T) = 0 \quad \text{in } \Omega_G \cup \Omega_S \\ \kappa \partial_n T = 0 \quad \text{on } \Gamma_N \\ T = T_D \quad \text{on } \Gamma_D \\ [\kappa \partial_n T] = 0 \quad \text{and } \gamma \\ T(0, x) = T_0 \quad \text{in } \Omega \end{array} \right.$$

The enthalpy  $H(T)$  is given by:

$$H(T) = \begin{cases} [(\rho c)_S^s(1 - f(T)) + (\rho c)_S^l f(T)]T + \rho_S \Delta h_{mf}(T) & \text{in } \Omega_S \\ (\rho c)_G T & \text{in } \Omega_G \end{cases}$$

# The time scheme

Implicit Euler :  $\frac{H^{n+1}-H^n}{\Delta t} - \text{div}(\kappa \nabla T^{n+1}) = 0$

Chernoff Scheme :  $H = \beta^{-1}(T)$  where  $T = \beta(H) \rightarrow \partial_t H = 1/\beta'(T) \partial_t T$

on définie:

$$H^{n+1} = H^n + \gamma(T^{n+1} - \beta(H^n))$$

where  $\gamma$  corresponds to  $\frac{1}{\beta'(T)}$

$$\frac{\gamma}{\Delta t}(T^{n+1} - \beta(H^n)) - \text{div}(\kappa \nabla T^{n+1}) = 0$$

## Linear Problem

$$\lambda E(\mathbf{u}) - \operatorname{div}(\kappa \nabla \mathbf{u}) = g \quad \text{dans} \quad \Omega_G \cup \Omega_S$$

$$\kappa \partial_n \mathbf{u} = 0 \quad \text{sur} \quad \Gamma_N$$

$$\mathbf{u} = u_D \quad \text{sur} \quad \Gamma_D$$

$$[\kappa \partial_n \mathbf{u}] = 0 \quad \text{et} \quad R(\kappa \partial_n \mathbf{u})_s = [\mathbf{u}] \quad \text{sur} \quad \gamma$$

## Sobolov spaces

$$H(\operatorname{div}, \Omega) := \{q \in L^2(\Omega)^2 : \operatorname{div} q \in L^2(\Omega)\}$$

$$H_{0,N}(\operatorname{div}, \Omega) := \{q \in H(\operatorname{div}, \Omega) : q \cdot n = 0 \quad \text{sur} \quad \Gamma_N\}$$

## Mixed Formulation

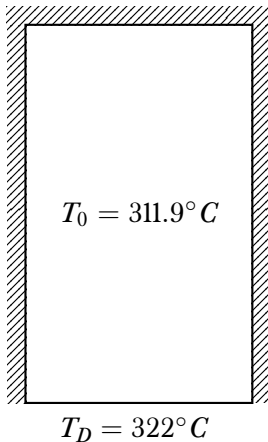
$$\mathbf{p} = \kappa \nabla \mathbf{u}$$

Find  $(\mathbf{u}, \mathbf{p})$  such as

$$\int_{\Omega} \frac{1}{\kappa} \mathbf{p} \cdot q \, dx + \int_{\Omega} \mathbf{u} \operatorname{div} q \, dx - \int_{\gamma} R(\mathbf{p} \cdot n)(q \cdot n) \, d\gamma = \int_{\Gamma_D} u_D q \cdot n \, d\gamma \quad \forall q \in H_{0,N}(\operatorname{div}, \Omega)$$

$$\int_{\Omega} \lambda E(\mathbf{u}) v \, dx - \int_{\Omega} v \operatorname{div} \mathbf{p} \, dx = \int_{\Omega} v g \, dx \quad \forall v \in L_2(\Omega)$$

# Initial and boundary conditions



# The analytic solution in the case of a homogeneous PCM

$$T(x, t) = T_D + (T_m - T_D) \frac{\operatorname{erf}\left(\frac{x}{2\sqrt{\alpha t}}\right)}{\operatorname{erf}(\xi)} \quad 0 < x \leq X(t)$$

with  $X(t) = 2\xi\sqrt{\alpha t}$  where  $\xi$  is solution to

$$\frac{\operatorname{Stefan}}{\operatorname{erf}(\xi)e^{\xi^2}} = \sqrt{\pi}\xi$$

with

$$\operatorname{Stefan} = \frac{c(T_D - T_m)}{\Delta h_m}$$

The heat flux at surface  $y = 0$  is given by

$$q(0, t) = -\frac{\kappa(T_m - T_D)}{\operatorname{erf}(\xi)\sqrt{\pi\alpha t}}$$

The similarity variable is defined by  $\mu = \frac{x}{\sqrt{\alpha t}}$

# PCM with equivalent thermo-physical properties

Recall : the porosity of a composite is defied as  $\mathcal{E} = \frac{V_{pores}}{V_{Total}}$

If  $\rho_G$  and  $\rho_S$  are the density of the matrix and the PCM then  $\rho_{equivalent} = \mathcal{E}\rho_S + (1 - \mathcal{E})\rho_G$

$C_{p_G}$  and  $C_{p_S}$  the thermal capacities of the matrix and of the PCM,

$$C_{p_{equivalent}} = \mathcal{E}C_{p_S} + (1 - \mathcal{E})C_{p_G}$$

$\Delta h_m$  is the latent heat of fusion of the PCM contained in the matrix,  $\Delta h_{m_{equivalent}} = \mathcal{E}\Delta h_m$

# Dimensionless Variables

$$x^* = \frac{x}{x_{max}}, \quad t^* = \frac{\alpha t}{x^2}, \quad T^* = \frac{T - T_D}{T_m - T_D}, \quad X^* = \frac{X}{x_{max}} \quad \text{and} \quad q^* = \frac{x_{max} q}{\kappa(T_m - T_D)}$$

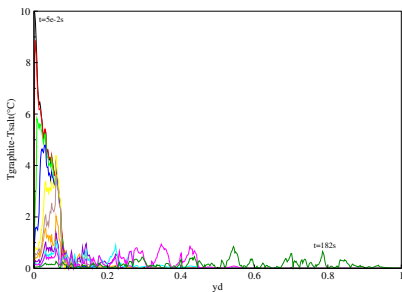
$$T^*(x^*, t) = \frac{\text{erf}\left(\frac{x^*}{2\sqrt{t^*}}\right)}{\text{erf}(\xi)}, \quad X^*(t) = 2\xi\sqrt{t^*}, \quad q^*(0, t) = -\frac{1}{\sqrt{\pi}\text{erf}(\xi)\sqrt{t^*}}$$

For a composite material, we introduce the mean dimensionless temperature

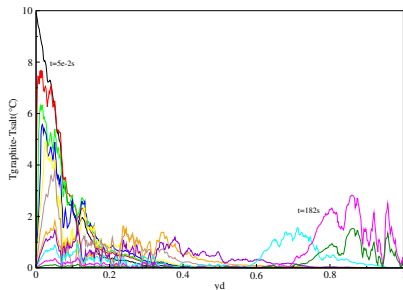
$$T_{Mean}^*(y^*, t) = \frac{1}{N_x} \sum_{n=1}^{n=N_x} T^*(x, y, t)$$

# Comparison with a homogeneous PCM

KL1\_250



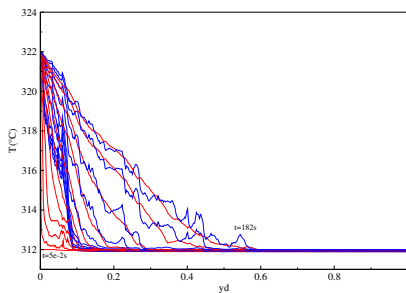
KD1



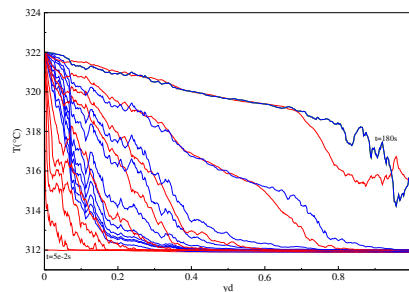


# KL1\_250: temperature profile

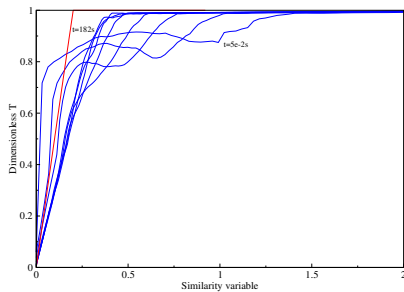
KL1\_250



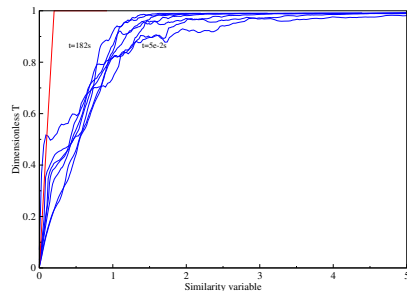
KD1



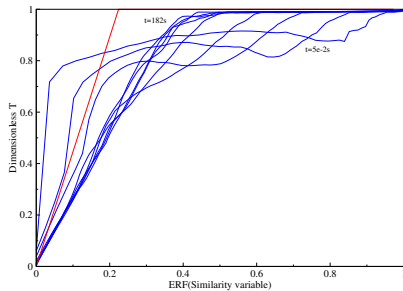
KL1\_250



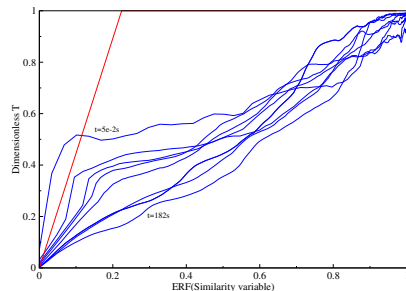
KD1

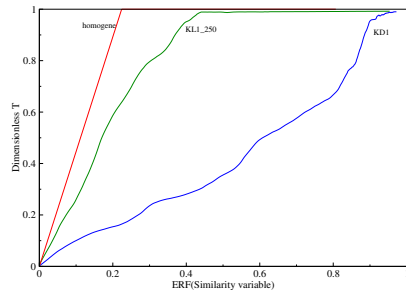
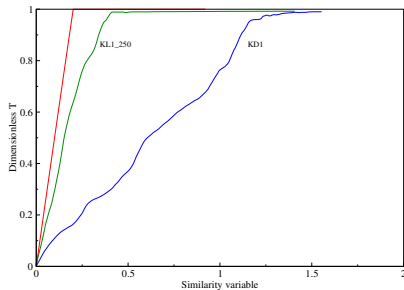


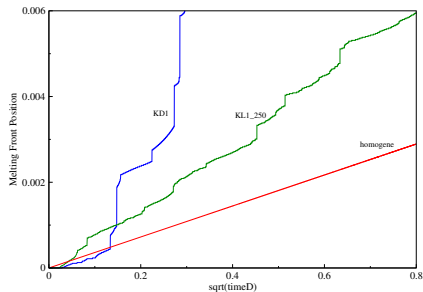
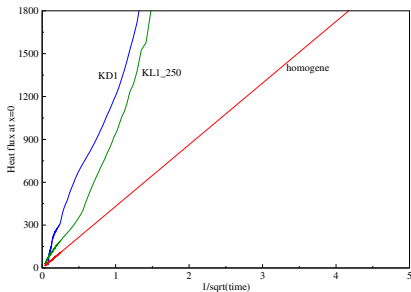
KL1\_250



KD1





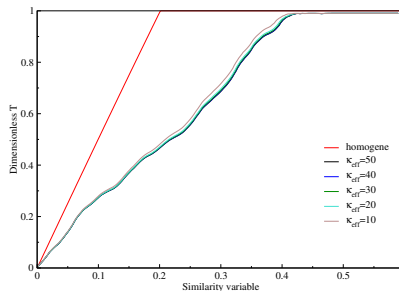


**Conclusion :** from  $t = 182s$  KL1\_250 could be assimilated to a homogeneous PCM with equivalent thermo-physical properties.

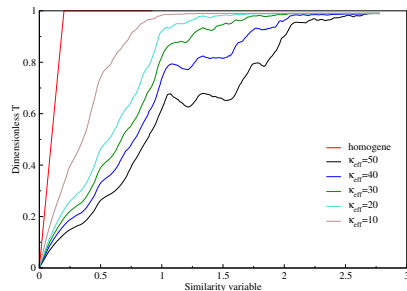
# Effects of increasing the effective thermal conductivity

$\kappa_{effective} (W/m/K)$	$\kappa_{Foam}$ KL1_250	$\kappa_{Foam}$ KD1
10	410	125
20	958	336
30	1510	580
40	2060	848
50	2615	1128

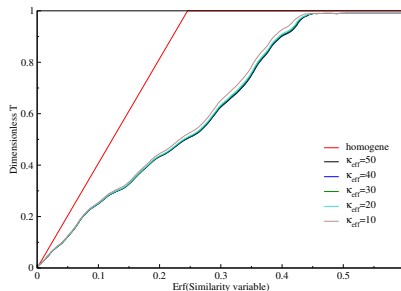
KL1\_250



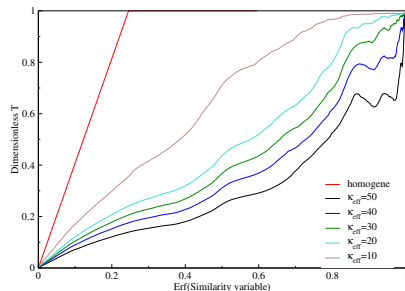
KD1



KL1\_250

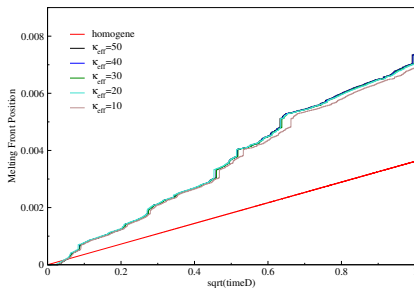


KD1

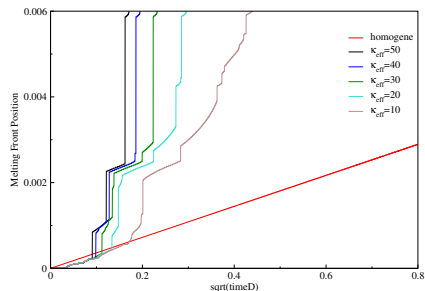




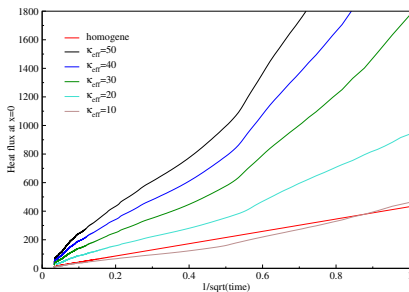
KL1\_250



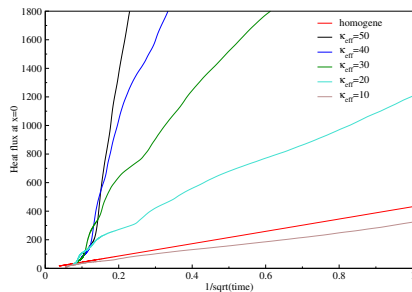
KD1

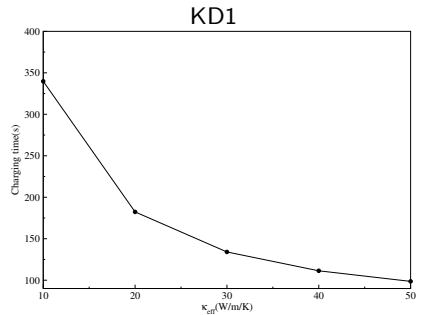
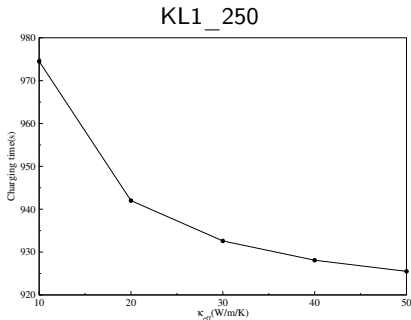


KL1\_250



KD1





## Conclusions :

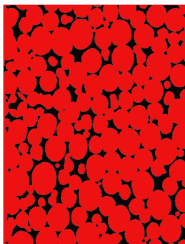
- Changing the values of the effective thermal conductivity has more effects in the case of KD1 because the volume occupied by the matrix represents 30% of the total volume where for KL1\_250 it represents only 18% of the total volume.
- Increasing the value of the effective thermal conductivity allows to accelerate the charging/discharging cycles.

# Influence of the pores size

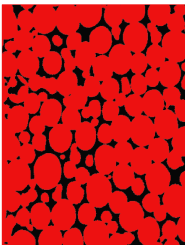
- Dilatation in both directions :

$$f_{new}(x, y) = f(\alpha x, \alpha y) \text{ with } \alpha \in \{0.6, 0.7, 0.8, 0.9, 1\}$$

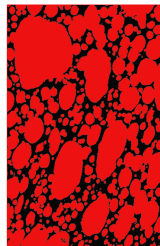
- $\kappa_{eff} = 20 \text{ W/m/K}, \forall \alpha$



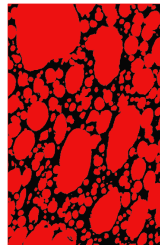
initial



$\alpha = 0.9$



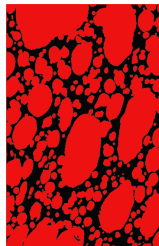
initial



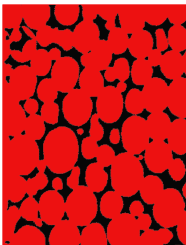
$\alpha = 0.9$



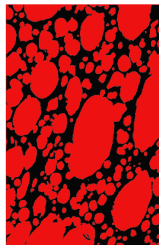
$\alpha = 0.8$



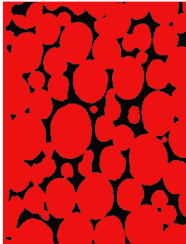
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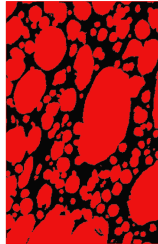
$\alpha = 0.7$



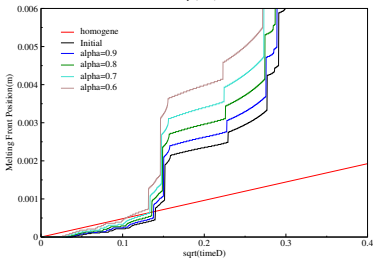
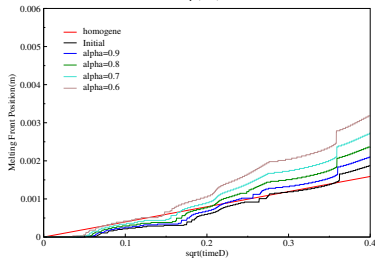
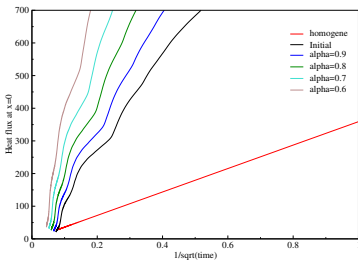
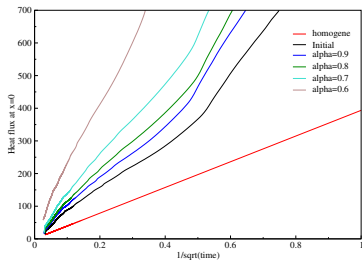
$\alpha = 0.7$



$$\alpha = 0.6$$



$$\alpha = 0.6$$





# Conclusions

Enlarging the pores allow to

- accelerate the melting process inside the ports.
- increasing the heat flux while keeping a constant thermal conductivity.

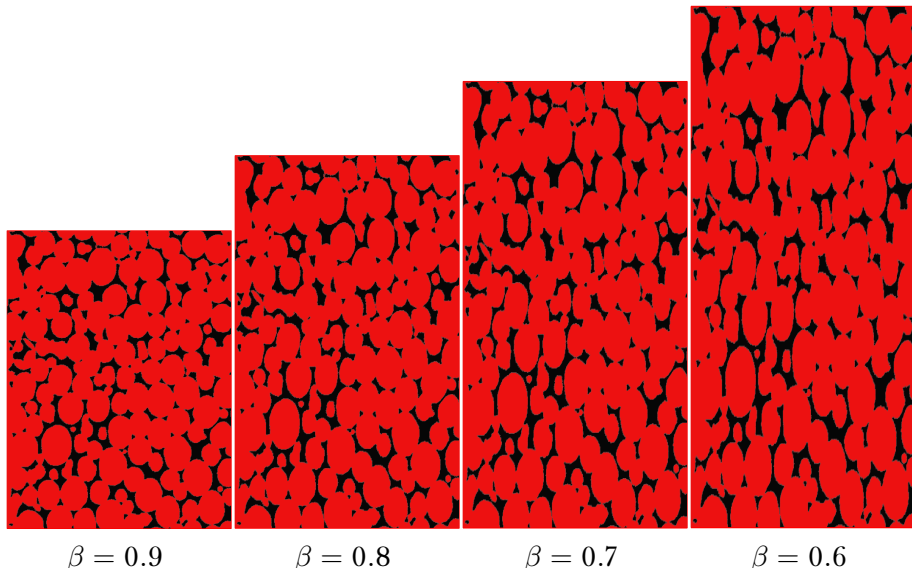
# Influence of the pores shape

- Dilatations in both directions :

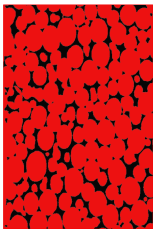
$$f_{nouveau}(x, y) = f(\alpha x, \beta y) \text{ avec } \alpha, \beta \in \{0.6, 0.7, 0.8, 0.9, 1\}$$

- $\kappa_{eff} = 20 \text{ W/m/K}, \forall \alpha, \beta$

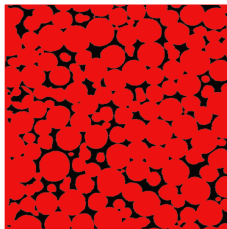
# Dilatations de KL1\_250 dans la direction $y$



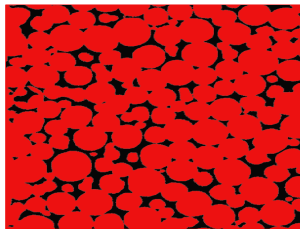
# Dilatations de KL1\_250 dans la direction $x$



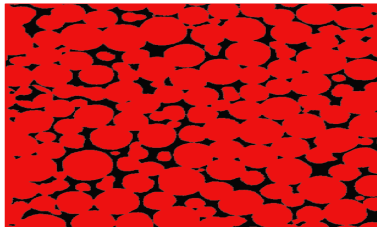
$\alpha = 0.9$



$\alpha = 0.8$

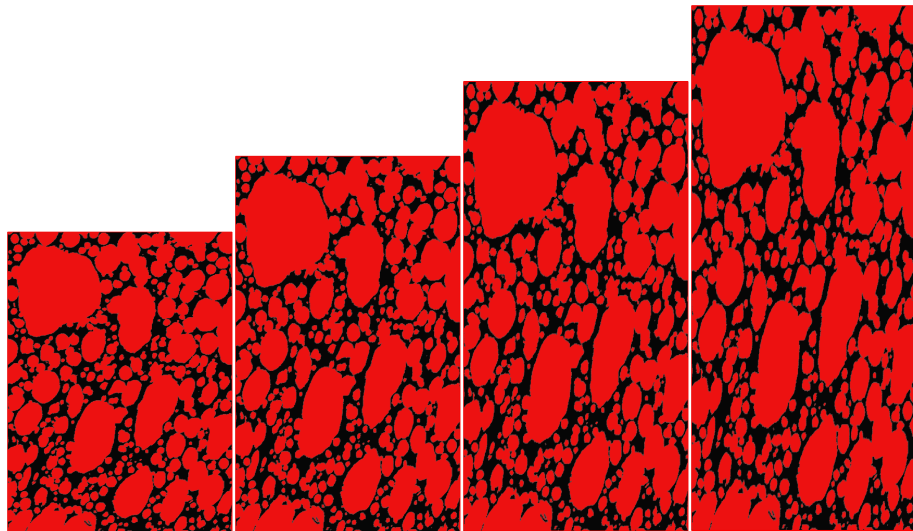


$\alpha = 0.7$



$\alpha = 0.6$

# Dilatations de KD1 dans la direction $y$



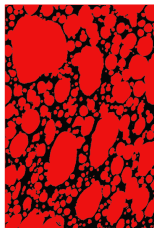
$\beta = 0.9$

$\beta = 0.8$

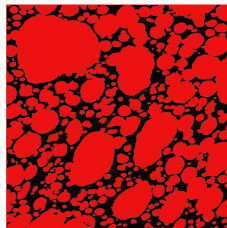
$\beta = 0.7$

$\beta = 0.6$

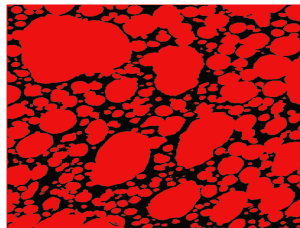
# Dilatations de KD1 dans la direction $x$



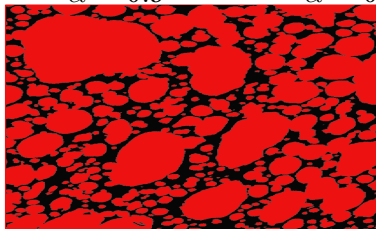
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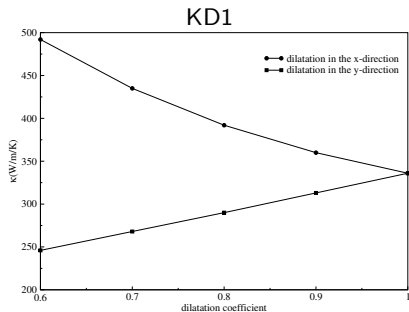
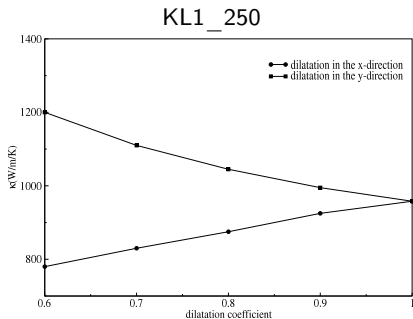
$\alpha = 0.8$



$\alpha = 0.7$



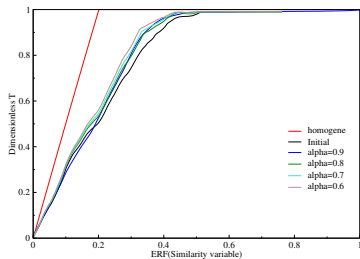
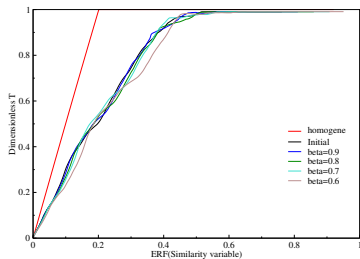
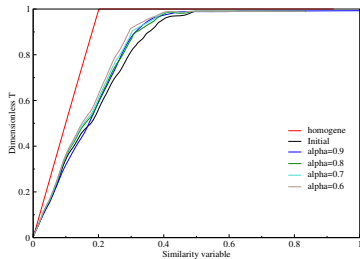
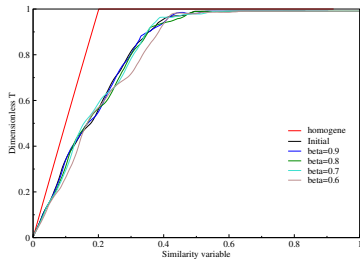
$\alpha = 0.6$



- For KL1\_250 we can see that the more we poll in the **y** direction the more we need to increase the thermal conductivity of the matrix and the more we poll in the **x** direction the more we increase the thermal conductivity of the matrix.
- On note the opposite for KD1.

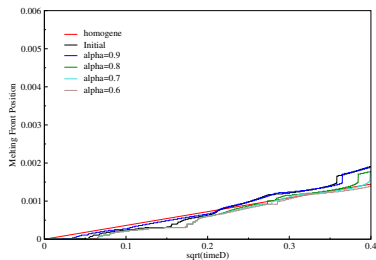
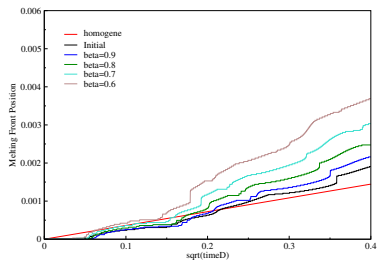
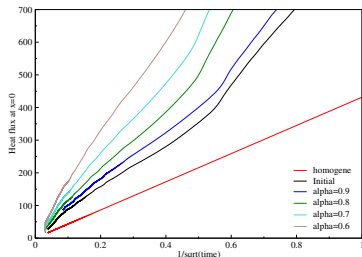
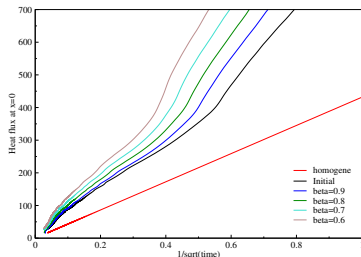
Explication : for KL1\_250 les pores s'étalent dans la direction **y** et on sollicite dans cette direction → on doit augmenter  $\kappa_{matrice}$ .

# Résultats pour KL1\_250

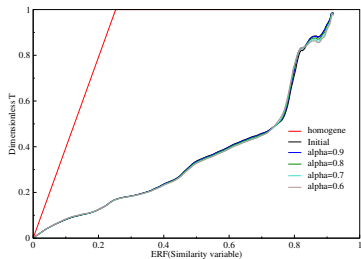
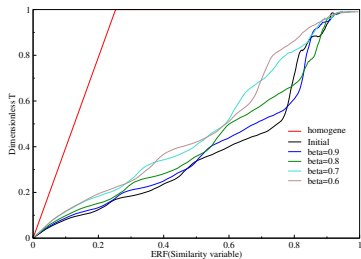
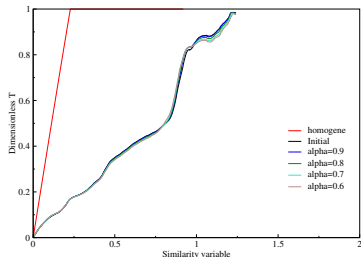
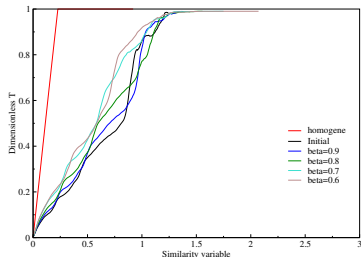




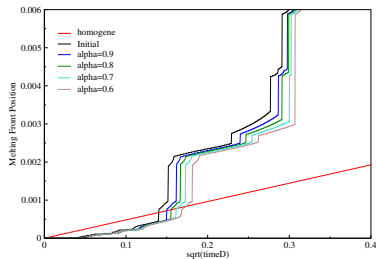
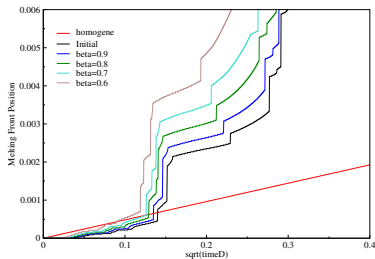
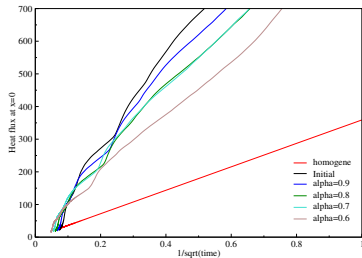
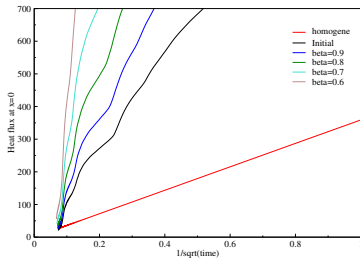
# Résultats pour KL1\_250



# Résultats pour KD1



# Résultats pour KD1



# Conclusions

En dilatant dans la direction  $x$

- Pour KL1\_250 on remarque des variations dans les profils de températures ainsi qu'aux flux surfaciques puisqu'on augmente la valeur de  $\kappa_{matrice}$ .
- Pour KD1 on a presque le même comportement pour les différents  $\alpha$  puisqu'on ne sollicite pas dans cette direction.

En dilatant dans la direction  $y$  on constate que dans les deux cas les fronts de fusions vont plus vite puisque les pores se sont allongés dans le sens où les matériaux sont "chauffés".

Donc pour accélérer les cycles de charges/décharges il faudrait dilater les pores dans le sens du gradient de température.

# Bibliographie



V. Canseco, Y. Anguy, J. J. Roa, and E. Palomo, *Structural and mechanical characterization of graphite foam/phase change material composites*, Carbon, 74 (2014), pp. 266–281.



R. M. Christensen, *Mechanics of composite materials*, Courier Corporation, 2012.



H. Jopek and T. Strek, *Optimization of the effective thermal conductivity of a composite*, INTECH Open Access Publisher, 2011.



A. L. Kalamkarov, I. V. Andrianov, V. V. Danishevsâ, et al., *Asymptotic homogenization of composite materials and structures*, Applied Mechanics Reviews, 62 (2009), p. 030802.



A. L. Kalamkarov and K. S. Challagulla, *Effective properties of composite materials, reinforced structures and smart composites: Asymptotic homogenization approach*, in Effective Properties of Heterogeneous Materials, Springer, 2013, pp. 283–363.



V. Morisson, *Heat transfer modelling within graphite/salt composites: from the pore scale equations to the energy storage system*, PhD thesis, Bordeaux 1, 2008.