Advanced Materials formula booklet

By: Ioannis Karras

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Atomic bonding (Lecture 1)

| Fraction of ionic | $f_{ion} = 1 - e^{-\frac{(X_A - X_B)^2}{4}}$ |
|-------------------|--|
| bonding | where X_A and X_B are the electronegativities of the two bonded elements |

Crystallography and structure of metals (Lecture 2)

| Atomic packing factor | $APF = \frac{\text{volume of atoms in a unit cell}}{\text{total unit cell volume}}$ Note that face atoms are counted as half atoms, and corner atoms are counted as eighth atoms |
|--|---|
| Unit cell edge length for body-centered cubic | $a = \frac{4R}{\sqrt{3}}$ |
| Unit cell edge length for face-centered cubic | $a = 2R\sqrt{2}$ |
| Density of a metal with a unit cell | $\rho \approx \frac{\text{mass of atoms in a unit cell}}{\text{total unit cell volume}} = \frac{nA}{V_C N_A}$ n is the number of atoms associated with each unit cell A is the atomic weight V_C is the volume of the unit cell |
| | N_A is Avogadro's number, $6.02 \cdot 10^{23} \text{ mol}^{-1}$ |
| Density of a metal in terms of atomic packing factor | $\rho \approx \mathit{APF} \cdot \rho_{atom} = \mathit{APF} \cdot \frac{m_{atom}}{V_{atom}}$ where m_{atom} is the mass of a single atom and V_{atom} is the volume of a single atom |
| Atomic packing factors for cubic lattices | sc: $APF = 0.52$ bcc: $APF = 0.68$ fcc: $APF = 0.74$ |

Lattice defects (Lecture 3)

| Equilibrium density of vacancies | $c_V = \frac{N_V}{N} = e^{-\left(\frac{Q_V}{kT}\right)}$ where N_V is the number of vacancies, N is the number of atomic sites, Q_V is the vacancy formation energy, and k is the Boltzmann constant, $1.38\cdot 10^{-23}$ J/K |
|--|--|
| Number of atoms per unit volume | $N = \frac{N_A \rho}{M}$ where N_A is Avogadro's number, $6.02 \cdot 10^{23} \ \rm mol^{-1}, \ \rho$ is density, and M is molar mass |
| Dislocation density | $\rho = \frac{L}{V}$ where L is dislocation line length and V is volume |
| Mass fraction/weight percentage for a binary alloy | $C_A = rac{m_A}{m_A + m_B}$ |
| Amount fraction/atomic percentage for a binary alloy | $C_{\!A}^* = rac{n_{\!A}}{n_{\!A} + n_{\!B}}$ where n is number of moles |
| Conversion between mass fraction and amount fraction | $C_A = \frac{C_A^* M_A}{C_A^* M_A + C_B^* M_B}$ $C_A^* = \frac{\frac{C_A}{M_A}}{\frac{C_A}{M_A} + \frac{C_B}{M_B}}$ where M is molar mass |

Ceramics and polymers (Lecture 4)

| Determination of | $\left \frac{r_{A+}}{r_{X-}} < 0.414$: Zinc blende structure |
|-------------------|--|
| | $0.414 < \frac{r_{A+}}{r_{X-}} < 0.732$: Sodium chloride structure |
| ceramic structure | $\left \frac{r_{A+}}{r_{X-}} > 0.732$: Cesium chloride structure |
| | r_{A+} is the radius of the cation and r_{X-} is the radius of the anion |

| Molar mass of polymer chain molecule | $M=nM_{unit}$ where n is the number of repeating units and M_{unit} is the molar mass |
|---|--|
| Grain molecule | of a single repeating unit |
| Number-weighted average molar mass | $\langle M \rangle_n = \sum x_i M_i$ |
| | where x_i is the fraction of all chains having molar mass M_i |
| | $DP = \frac{M}{M_{mon}}$ |
| | where ${\it M}$ is the molar mass of the chain and ${\it M}_{mon}$ is the molar mass of a monomer |
| | For co-polymers: |
| Degree of polymerization (number of monomers in a | $DP = \frac{M}{\langle M_{mon} \rangle}$ |
| chain) | where $\langle M_{mon} \rangle$ is the average molar mass of the monomers |
| | For a distribution of chain lengths: |
| | $\langle DP \rangle = \frac{\langle M \rangle_n}{M_{mon}} \text{ or } \langle DP \rangle = \frac{\langle M \rangle_n}{\langle M_{mon} \rangle}$ |
| | where the angle brackets denote an average. $\langle M \rangle_n$ is the <i>number-weighted</i> average molar mass |
| Straightened chain | $L = Nd \sin\left(\frac{\theta}{2}\right)$ |
| length | where N is the number of single bonds along the chain, d is the bond length, and θ is the bond angle |
| Average start to end | $r = d\sqrt{N}$ |
| distance for a polymer chain | where d is the bond length and N is the number of single bonds along the chain |
| | $\rho_{sc} = f_c \rho_c + (1 - f_c) \rho_a$ |
| Density of a semicrystalline polymer | where ρ_c is the density of the crystalline region, ρ_a is the density of the amorphous region, and f_c is the volume fraction of crystalline regions. Transposing for f_c , |
| | $f_c = \frac{\rho_{sc} - \rho_a}{\rho_c - \rho_a}$ |
| Crystallinity | $x = \frac{m_c}{m_{sc}} = \frac{f_c \rho_c}{\rho_{sc}} = \frac{\rho_c (\rho_{sc} - \rho_a)}{\rho_{sc} (\rho_c - \rho_a)}$ |
| | where c is the crystalline region, a is the amorphous region, and sc is the whole semicrystalline polymer |

Imaging (Lecture 5)

| Interplanar spacing for a cubic crystal | $d_{hkl}=\frac{a}{\sqrt{h^2+k^2+l^2}}$ where a is the unit cell edge length and $h,k,$ and l are the Miller indices of the plane being considered |
|---|---|
| Bragg's law | $n\lambda = 2d_{hkl}\sin(\theta)$ n is the order of reflection, a positive integer λ is the x-ray wavelength 2θ , not θ , is the diffraction angle |
| Lens equation | $\frac{1}{d_i} + \frac{1}{d_0} = \frac{1}{f}$ where d_i is the distance from the lens to the image plane, d_0 is the distance from the lens to the object plane, and f is the focal length |
| Lens magnification | $M = \frac{d_i}{d_0}$ |
| Mean intercept length for grain size | Draw many random lines. The mean intercept length is $\bar{l} = \frac{L_T}{PM}$ where L_T is the total length of all the lines, P is the total number of intercepts, and M is the magnification |

Mechanical testing (Lecture 6)

| Tensile/compressive stress | $\sigma = \frac{F_t}{A_0}$ where F_t is the force applied and A_0 is the initial cross sectional area of the specimen. $\sigma > 0$ for tensile stress and $\sigma < 0$ for compressive stress |
|----------------------------|--|
| Tensile/compressive strain | $\varepsilon = \frac{\Delta l}{l_0}$ where Δl is the elongation and l_0 is the initial specimen length |
| Hooke's law | $\sigma = E \varepsilon$ where E is the elastic modulus, or Young's modulus |
| Poisson's ratio | $\nu=-\frac{\Delta d/d_0}{\Delta l/l_0}=-\frac{\varepsilon_x}{\varepsilon_z}$ where d is the length perpendicular to a load and l is the length parallel to it |

| Shear stress | $\tau = \frac{F_s}{A_0}$ |
|---|--|
| Shear strain | $\gamma = \frac{\Delta x}{h_0}$ where Δx is the displacement of one side and h_0 is the initial height |
| Hooke's law for shear | $	au = G \gamma$ where G is the shear modulus |
| Relationship between moduli for elastic isotropic materials | $G = \frac{E}{2(1+\nu)}$ |

Electrical and thermal properties (Lecture 7)

| Resistance (Ohm's law) | $R = \frac{V}{I}$ |
|--|---|
| | Unit: Ω . V is voltage and I is current |
| Resistivity | $\rho = \frac{RA}{L}$ |
| | Unit: $\Omega m.~A$ is cross-sectional area and L is length |
| Conductance | $C = \frac{1}{R}$ |
| | Unit: Ω^{-1} |
| Conductivity | $\sigma = \frac{1}{\rho}$ |
| | Unit: $(\Omega m)^{-1}$ |
| Current density | $J = \frac{I}{A}$ |
| · | Unit: A/m^2 . A is cross-sectional area |
| Electric field intensity | $\mathcal{E} = \frac{V}{L}$ |
| | Unit: V/m . L is length |
| Current density in terms of electron density | $J = n e v_d$ |
| | where n is the electron density, $ e $ is the absolute value of electron charge, $1.6\cdot 10^{-19}$ C, and v_d is the net drift velocity |
| Net drift velocity | $v_d = \mu_e \mathcal{E}$ |
| | where μ_e is the electron mobility |

| Contributions to resistivity are additive | Total resistivity is the sum of resistivities due to temperature, impurities, and dislocations |
|---|---|
| | $ ho_{total} = ho_{temp} + ho_{imp} + ho_{disloc}$ |
| Intrinsic conductivity | $\sigma = n_i e (\mu_e + \mu_h)$ |
| | where n_i is the intrinsic carrier concentration, equal to both the electron concentration n and the hole concentration p . μ_e is electron mobility and μ_h is hole mobility |
| | For n-type: |
| | $\sigma = n e \mu_e$ |
| | For p-type: |
| Extrinsic conductivity | $\sigma = p e \mu_h$ |
| | n is electron concentration, p is hole concentration, μ_e and μ_h are electron and hole mobilities respectively, and $ e $ is the absolute value of electron charge, $1.6 \cdot 10^{-19}$ C |
| | $Q = mc\Delta T$ |
| Heat required for a | $Q = nC\Delta T$ |
| Heat required for a temperature change | where c is specific heat capacity (per mass) and c is molar heat capacity. Heat capacities can be fixed-volume (c_v or c_v) or fixed-pressure (c_p or c_p) |
| Dependence of fixed- | $C_v = AT^3$ |
| volume molar heat | where A is a constant |
| capacity on temperature at low temperatures | For solid metallic elements, above the Debye temperature θ_D , C_v levels off at approximately $3R$, where R is the gas constant, 8.314J/(mol K) |
| | Linear: |
| | $\frac{\Delta L}{L_0} = \alpha_l \Delta T$ |
| | where ΔL is the increase in length, L_0 is the initial length, and α_l is the linear expansion coefficient |
| Thermal expansion | Volume: |
| | $\frac{\Delta V}{V_0} = \alpha_v \Delta T$ |
| | where ΔV is the increase in volume, V_0 is the initial volume, and α_v is the volume expansion coefficient |
| | $\sigma = E \alpha_l \Delta T$ |
| Thermal stress | where α_l is the linear expansion coefficient and ${\it E}$ is the elastic modulus |

| Thermal shock resistance | $TSR \approx \frac{\sigma_f k}{E \alpha_l}$ where σ_f is fracture strength, k is thermal conductivity, E is the elastic modulus, and α_l is the linear expansion coefficient |
|---|--|
| Heat flux in steady state heat flow | $q = \frac{\frac{dQ}{dt}}{A} = \frac{H}{A} = \frac{k(T_H - T_C)}{L}$ H is heat current (power) A is the area through which the heat flows k is thermal conductivity T_H is the temperature of the hot end T_C is the temperature of the cold end L is the length of the heat flow path |
| Contributions to thermal conductivity are additive | Total conductivity is the sum of conductivities due to the lattice and due to free electrons $k=k_{lattice}+k_{electron}$ |
| Relationship between electrical and thermal conductivity for metals (Wiedemann-Franz law) | $L = \frac{k}{\sigma T}$ where L is a constant which is approximately the same for all metals. k is thermal conductivity and σ is electrical conductivity. The theoretical value of L is $2.44 \cdot 10^{-8} \; \Omega \text{W/K}^2$ |

Phase diagrams (Lecture 9)

| Mass fraction of each phase in a two-phase region | $W_A = \frac{\text{length from the point to the other phase on the tie line}}{\text{total length of the tie line}} = \frac{ c_B - c_0 }{ c_B - c_A }$ |
|---|---|
| Gibbs phase rule | number of degrees of freedom + number of phases in equilibrium = number of constituents + 1 |

| | eutectic $L \rightarrow S_1 + S_2 \qquad \frac{L}{S_1 + S_2}$ |
|-------------------------------|---|
| Special phase transformations | eutectoid $S_1 \rightarrow S_2 + S_3$ $S_2 + S_3$ |
| | peritectic $S_1 + L \rightarrow S_2$ S_2 |

Plastic deformation and strengthening (Lecture 10)

| Theoretical yield strength for a defect-free material | $\sigma_{th} = \frac{E}{20}$ | |
|--|--|--|
| Theoretical critical shear stress | $\tau_{th} = \frac{G}{20}$ | |
| Percent cold work | $\%CW = \frac{A_0 - A_d}{A_0} \cdot 100\%$ where A_0 is the original area of the cross-section that experiences deformation and A_d is the cross-sectional area after deformation | |
| Total strength due to different factors is additive | $\sigma = \sigma_0 + \Delta \sigma_{ss} + \Delta \sigma_{disl} + \Delta \sigma_{gb} + \Delta \sigma_p$ | |
| Dependence of yield strength on work hardening | $\sigma_y = \sigma_0 + \Delta \sigma_{disl} = \sigma_0 + M\alpha Gb\sqrt{\rho}$ where M is the Taylor factor, α is the interaction coefficient, G is the shear modulus, and b is the length of the Burgers vector | |
| Dependence of yield strength on grain size (Hall-Petch relation) | n size | |

| | Coherent: |
|--|--|
| Contribution to yield strength from particle strengthening | $\Delta \sigma_{ m p, coh} \propto rac{\gamma r}{I+2r}$ |
| | where γ is the interface energy, I is the distance between particles, and r is the particle radius |
| | Incoherent: |
| | $\Delta \sigma_{ m p,inchoh} \propto \frac{Gb}{I}$ |
| | where G is the shear modulus, b is the length of the Burgers vector, and I is the distance between particles |
| Relation between Brinell and Vickers hardness | $HB \approx 0.95 HV$ |
| Relation between | $UTS \approx c_{emp}HB$ |
| ultimate tensile strength and Brinell hardness | where c_{emp} is a constant that depends on material type |

Failure (Lecture 11)

| Maximum stress at a crack tip | $\sigma_{max}\approx 2\sigma_0\sqrt{\frac{a}{\rho_t}}$ where σ_0 is the applied stress, ρ_t is the radius of curvature of the crack tip, and a is either the length of a surface crack or half the length of an internal crack | |
|---|--|--|
| Critical stress for crack propagation in a brittle material | $\sigma_c = \sqrt{\frac{2E\gamma_s}{\pi a}}$ where E is the elastic modulus, γ_s is the specific surface energy, and a is either the length of a surface crack or half the length of an internal crack | |
| Fracture toughness | $K_{Ic} = Y\sigma_c\sqrt{\pi a}$ where Y is a parameter approximately equal to 1 for thick specimens, σ_c is the critical stress above, and a is either the length of a surface crack or half the length of an internal crack | |
| Stress intensity factor | $K_I = Y\sigma\sqrt{\pi a}$ as for fracture toughness, but for any stress. Catastrophic crack growth occurs when $K_I \geq K_{Ic}$ | |

| | Rectangular cross-section: | |
|-------------------|---|--|
| Flexural strength | $\sigma_{fs} = \frac{3F_fL}{2bd^2}$ where F_f is the load at fracture, L is the distance between support points, and b and d are as indicated in the figure | Possible cross sections F d Rectangular Circular |
| | Circular cross-section: σ_{p} where R is the radius of the spec | $f_{fs} = rac{F_f L}{\pi R^3}$ cimen |

Composites and material indices (Lecture 13)

| Longitudinal elastic modulus for a fiber/lamellar composite or upper bound for all composites | $E_{c\parallel}=f_1E_1+f_2E_2$ where f is volume fraction, E is elastic modulus, and the subscripts represent different phases. This situation is iso-strain |
|--|--|
| Ratio of load carried by fibers and the matrix phase for longitudinal loading | $\frac{F_f}{F_m} = \frac{E_f f_f}{E_m f_m}$ where f is volume fraction, E is elastic modulus, and the subscripts represent different phases |
| Transverse elastic modulus for a fiber/lamellar composite or lower bound for all composites | $\frac{1}{E_{c\perp}} = \frac{f_1}{E_1} + \frac{f_2}{E_2}$ where f is volume fraction, E is elastic modulus, and the subscripts represent different phases. This situation is iso-stress |
| Density of a composite | $\rho_c = f_1 \rho_1 + f_2 \rho_2$ where f is volume fraction and the subscripts represent different phases |
| General tensile stress for a fiber composite with longitudinal applied stress | $\sigma(\varepsilon) = f_m \sigma_m(\varepsilon) + f_f \sigma_f(\varepsilon) = f_m \sigma_m(\varepsilon) + f_f E_f \varepsilon$ where f is volume fraction, ε is strain, $\sigma(\varepsilon)$ is the stress of a phase at a certain strain, and E is elastic modulus. The subscript m represents the matrix and the subscript f represents the fibers |

| Violal atmospheric of a fibour | $\sigma_{\rm y,c} = f_m \sigma_{\rm y,m} + f_f E_f \varepsilon_{\rm y,m}$ | | | | |
|--|---|------------|-----------------------|--|--|
| Yield strength of a fiber composite with longitudinal applied stress | f_m and f_f are the volume fractions of the matrix and fibers respectively $\sigma_{\rm v,m}$ is the yield strength of the matrix | | | | |
| | E_f is the elastic modulus of the fibers | | | | |
| | $arepsilon_{ m y,m}$ is the yield strain of the matrix | | | | |
| Tensile strength of a fiber composite with | $\sigma^* = f_m \sigma_m' + f_f \sigma_f^*$ $f_m \text{ and } f_f \text{ are the volume fractions of the matrix and fibers respectively}$ | | | | |
| longitudinal applied stress | σ'_m is the stress on the matrix at the failure strain of the fibers σ^*_f is the fracture strength of the fibers | | | | |
| | Member | | Loading | Index | |
| | Beam | Stiffness | Tension / compression | E/ρ | |
| | | | Torsion | G/ ho | |
| | | | Bending | $E^{1/3}/\rho E^{1/2}/\rho$ | |
| | | Buckling | Compression | ${\it E}^{\scriptscriptstyle 1/2}/ ho$ | |
| Material indices for | Panel | Stiffness | Bending | $E^{1/3}/\rho$ | |
| lightweight structures | | | | | |
| | Beam | Strength* | Tension / compression | $\sigma_{_{_{V}}}/\rho$ | |
| | | Strength* | Bending | $\sigma_{_{V}}^{^{2/3}}/\rho$ | |
| | Panel | Strength* | Bending | $\sigma_{_{_{_{\hspace{-0.05cm}V}}}}^{^{1/2}}/ ho$ | |
| | Spring | Resilience | | $\sigma_y^2/E\rho$ | |
| | * Either yield strength or failure strength | | | | |