

Material Models and the Uintah Computational Framework

July 30, 2008

Contents

1	Introduction	2
2	Load Curves	2
3	Stress Update Algorithms	4
3.1	Simplified theory for hypoelastic-plasticity	5
3.2	Models	7
3.3	Equation of State Models	7
3.3.1	Melting Temperature	9
3.3.2	Shear Modulus	11
3.3.3	Flow Stress	12
3.3.4	Adiabatic Heating and Specific Heat	17
3.4	Adding new models	18
3.5	Damage Models and Failure	20
3.5.1	Yield conditions	20
3.5.2	Porosity model	21
3.5.3	Damage model	22
3.6	Erosion algorithm	22
3.7	Implementation	24
4	Example Input Files	25
4.1	Hypoelastic-plastic model	25

4.2 Elastic-plastic model	27
4.2.1 An exploding ring experiment	30

1 Introduction

This document deals with some features of Uintah and some material models for solids that have been implemented in the Uintah Computational Framework (for use with the Material Point Method). The approach taken has been to separate the stress-strain relations from the numerical stress update algorithms as far as possible. A moderate rotation/small strain hypoelastic-plastic stress update algorithm is discussed and the manner in which plasticity flow rules, damage models and equations of state fit into the stress update algorithms are shown.

2 Load Curves

It is often more convenient to apply a specified load at the MPM particles. The load may be a function of time. Such a load versus time curve is called a **load curve**.

In Uintah, the load curve infrastructure is available for general use (and not only for particles). However, it has been implemented only for a special case of pressure loading.

We invoke the load curve in the <MPM> section of the input file using `<use_load_curves> true </use_load_curves>`. The default value is `<use_load_curves> false </use_load_curves>`.

In Uintah, a load curve infrastructure is implemented in the file `.../MPM/PhysicalBC/LoadCurve.h`. This file is essentially a templated structure that has the following private data

```
// Load curve information
std::vector<double> d_time;
std::vector<T> d_load;
int d_id;
```

The variable `d_id` is the load curve ID, `d_time` is the time, and `d_load` is the load. Note that the load can have any form - scalar, vector, matrix, etc.

In our current implementation, the actual specification of the load curve information is in the <PhysicalBC> section of the input file. The implementation is limited in that it applies only to pressure boundary conditions for some special geometries (the implementation is in `.../MPM/PhysicalBC/PressureBC.cc`). However, the load curve template can be used in other, more general, contexts.

A sample input file specification of a pressure load curve is shown below. In this case, a pressure is applied to the inside and outside of a cylinder. The pressure is ramped up from 0 to 1 GPa on the inside and from 0 to 0.1 MPa on the outside over a time of 10 microsecs.

```

<PhysicalBC>
  <MPM>
    <pressure>
      <geom_object>
        <cylinder label = "inner cylinder">
          <bottom>          [0.0,0.0,0.0]    </bottom>
          <top>             [0.0,0.0,.02]    </top>
          <radius>          0.5              </radius>
        </cylinder>
      </geom_object>
      <load_curve>
        <id>1</id>
        <time_point>
          <time> 0 </time>
          <load> 0 </load>
        </time_point>
        <time_point>
          <time> 1.0e-5 </time>
          <load> 1.0e9 </load>
        </time_point>
      </load_curve>
    </pressure>
    <pressure>
      <geom_object>
        <cylinder label = "outer cylinder">
          <bottom>          [0.0,0.0,0.0]    </bottom>
          <top>             [0.0,0.0,.02]    </top>
          <radius>          1.0              </radius>
        </cylinder>
      </geom_object>
      <load_curve>
        <id>2</id>
        <time_point>
          <time> 0 </time>
          <load> 0 </load>
        </time_point>
        <time_point>
          <time> 1.0e-5 </time>
          <load> 101325.0 </load>
        </time_point>
      </load_curve>
    </pressure>
  </MPM>
</PhysicalBC>

```

The complete input file can be found in
 Uintah/StandAlone/inputs/MPM/thickCylinderMPM.ups.

3 Stress Update Algorithms

The hypoelastic-plastic stress update is based on an additive decomposition of the rate of deformation tensor into elastic and plastic parts while the hyperelastic-plastic stress update is based on a multiplicative decomposition of the elastic and plastic deformation gradients. Incompressibility is assumed for plastic deformations. The volumetric response is therefore determined either by a bulk modulus and the trace of the rate of deformation tensor or using an equation of state. The deviatoric response is determined either by an elastic constitutive equation or using a plastic flow rule in combination with a yield condition.

The material models that can be varied in these stress update approaches are (not all are applicable to both hypo- and hyperelastic formulations nor is the list exhaustive):

1. The elasticity model, for example,
 - Isotropic linear elastic model.
 - Anisotropic linear elastic models.
 - Isotropic nonlinear elastic models.
 - Anisotropic nonlinear elastic models.
2. Isotropic hardening or Kinematic hardening using a back stress evolution rule, for example,
 - Ziegler evolution rule .
3. The flow rule and hardening/softening law, for example,
 - Perfect plasticity/power law hardening plasticity.
 - Johnson-Cook plasticity .
 - Mechanical Threshold Stress (MTS) plasticity .
 - Anand plasticity .
4. The yield condition, for example,
 - von Mises yield condition.
 - Drucker-Prager yield condition.
 - Mohr-Coulomb yield condition.
5. A continuum or nonlocal damage model with damage evolution given by, for example,
 - Johnson-Cook damage model.
 - Gurson-Needleman-Tvergaard model.
 - Sandia damage model.
6. An equation of state to determine the pressure (or volumetric response), for example,
 - Mie-Gruneisen equation of state.

The currently implemented stress update algorithms in the Uintah Computational Framework do not allow for arbitrary elasticity models, kinematic hardening, arbitrary yield conditions and continuum or nonlocal damage (however the a damage parameter is updated and used in the erosion algorithm). The models that can be varied are the flow rule models, damage variable evolution models and the equation of state models.

Note that there are no checks in the Uintah Computational Framework to prevent users from mixing and matching inappropriate models.

This section describes the current implementation of the hypoelastic- plastic model. The stress update algorithm is a slightly modified version of the approach taken by Nemat-Nasser et al. (1991,1992) [1, 2], Wang (1994) [3], Maudlin (1996) [4], and Zocher et al. (2000) [5].

3.1 Simplified theory for hypoelastic-plasticity

A simplified version of the theory behind the stress update algorithm (in the context of von Mises plasticity) is given below.

Following [4], the rotated spatial rate of deformation tensor (\mathbf{d}) is decomposed into an elastic part (\mathbf{d}^e) and a plastic part (\mathbf{d}^p)

$$\mathbf{d} = \mathbf{d}^e + \mathbf{d}^p \quad (1)$$

If we assume plastic incompressibility ($\text{tr}(\mathbf{d}^p) = 0$), we get

$$\boldsymbol{\eta} = \boldsymbol{\eta}^e + \boldsymbol{\eta}^p \quad (2)$$

where $\boldsymbol{\eta}$, $\boldsymbol{\eta}^e$, and $\boldsymbol{\eta}^p$ are the deviatoric parts of \mathbf{d} , \mathbf{d}^e , and \mathbf{d}^p , respectively. For isotropic materials, the hypoelastic constitutive equation for deviatoric stress is

$$\dot{\mathbf{s}} = 2\mu(\boldsymbol{\eta} - \boldsymbol{\eta}^p) \quad (3)$$

where \mathbf{s} is the deviatoric part of the stress tensor and μ is the shear modulus. We assume that the flow stress obeys the Huber-von Mises yield condition

$$f := \sqrt{\frac{3}{2}} \|\mathbf{s}\| - \sigma_y \leq 0 \quad \text{or} \quad F := \frac{3}{2} \mathbf{s} : \mathbf{s} - \sigma_y^2 \leq 0 \quad (4)$$

where σ_y is the flow stress. Assuming an associated flow rule, and noting that $\mathbf{d}^p = \boldsymbol{\eta}^p$, we have

$$\boldsymbol{\eta}^p = \mathbf{d}^p = \lambda \frac{\partial f}{\partial \boldsymbol{\sigma}} = \Lambda \frac{\partial F}{\partial \boldsymbol{\sigma}} = 3\Lambda \mathbf{s} \quad (5)$$

where $\boldsymbol{\sigma}$ is the stress. Let \mathbf{u} be a tensor proportional to the plastic straining direction, and define γ as

$$\mathbf{u} = \sqrt{3} \frac{\mathbf{s}}{\|\mathbf{s}\|}; \quad \gamma := \sqrt{3}\Lambda \|\mathbf{s}\| \quad \implies \gamma \mathbf{u} = 3\Lambda \mathbf{s} \quad (6)$$

Therefore, we have

$$\boldsymbol{\eta}^p = \gamma \mathbf{u}; \quad \dot{\mathbf{s}} = 2\mu(\boldsymbol{\eta} - \gamma \mathbf{u}) \quad (7)$$

From the consistency condition, if we assume that the deviatoric stress remains constant over a timestep, we get

$$\gamma = \frac{\mathbf{s} : \boldsymbol{\eta}}{\mathbf{s} : \mathbf{u}} \quad (8)$$

which provides an initial estimate of the plastic strain-rate. To obtain a semi-implicit update of the stress using equation (7), we define

$$\tau^2 := \frac{3}{2} \mathbf{s} : \mathbf{s} = \sigma_y^2 \quad (9)$$

Taking a time derivative of equation (9) gives us

$$\sqrt{2}\dot{\tau} = \sqrt{3} \frac{\mathbf{s} : \dot{\mathbf{s}}}{\|\mathbf{s}\|} \quad (10)$$

Plugging equation (10) into equation (7)₂ we get

$$\dot{\tau} = \sqrt{2}\mu(\mathbf{u} : \boldsymbol{\eta} - \gamma \mathbf{u} : \mathbf{u}) = \sqrt{2}\mu(d - 3\gamma) \quad (11)$$

where $d = \mathbf{u} : \boldsymbol{\eta}$. If the initial estimate of the plastic strain-rate is that all of the deviatoric strain-rate is plastic, then we get an approximation to γ , and the corresponding error (γ_{er}) given by

$$\gamma_{\text{approx}} = \frac{d}{3}; \quad \gamma_{\text{er}} = \gamma_{\text{approx}} - \gamma = \frac{d}{3} - \gamma \quad (12)$$

The incremental form of the above equation is

$$\Delta\gamma = \frac{d^* \Delta t}{3} - \Delta\gamma_{\text{er}} \quad (13)$$

Integrating equation (11) from time t_n to time $t_{n+1} = t_n + \Delta t$, and using equation (13) we get

$$\tau_{n+1} = \tau_n + \sqrt{2}\mu(d^* \Delta t - 3\Delta\gamma) = \tau_n + 3\sqrt{2}\mu\Delta\gamma_{\text{er}} \quad (14)$$

where d^* is the average value of d over the timestep. Solving for $\Delta\gamma_{\text{er}}$ gives

$$\Delta\gamma_{\text{er}} = \frac{\tau_{n+1} - \tau_n}{3\sqrt{2}\mu} = \frac{\sqrt{2}\sigma_y - \sqrt{3}\|\mathbf{s}_n\|}{6\mu} \quad (15)$$

The direction of the total strain-rate (\mathbf{u}^η) and the direction of the plastic strain-rate (\mathbf{u}^s) are given by

$$\mathbf{u}^\eta = \frac{\boldsymbol{\eta}}{\|\boldsymbol{\eta}\|}; \quad \mathbf{u}^s = \frac{\mathbf{s}}{\|\mathbf{s}\|} \quad (16)$$

Let θ be the fraction of the time increment that sees elastic straining. Then

$$\theta = \frac{d^* - 3\gamma_n}{d^*} \quad (17)$$

where $\gamma_n = d_n/3$ is the value of γ at the beginning of the timestep. We also assume that

$$d^* = \sqrt{3}\boldsymbol{\eta} : [(1 - \theta)\mathbf{u}^\eta + \frac{\theta}{2}(\mathbf{u}^\eta + \mathbf{u}^s)] \quad (18)$$

Plugging equation (17) into equation (18) we get a quadratic equation that can be solved for d^* as follows

$$\frac{2}{\sqrt{3}}(d^*)^2 - (\boldsymbol{\eta} : \mathbf{u}^s + \|\boldsymbol{\eta}\|)d^* + 3\gamma_n(\boldsymbol{\eta} : \mathbf{u}^s - \|\boldsymbol{\eta}\|) = 0 \quad (19)$$

The real positive root of the above quadratic equation is taken as the estimate for d . The value of $\Delta\gamma$ can now be calculated using equations (13) and (15). A semi-implicit estimate of the deviatoric stress can be obtained at this stage by integrating equation (7)₂

$$\tilde{s}_{n+1} = s_n + 2\mu \left(\eta\Delta t - \sqrt{3}\Delta\gamma \frac{\tilde{s}_{n+1}}{\|\tilde{s}_{n+1}\|} \right) \quad (20)$$

$$= s_n + 2\mu \left(\eta\Delta t - \frac{3}{\sqrt{2}}\Delta\gamma \frac{\tilde{s}_{n+1}}{\sigma_y} \right) \quad (21)$$

Solving for \tilde{s}_{n+1} , we get

$$\tilde{s}_{n+1} = \frac{s_{n+1}^{\text{trial}}}{1 + 3\sqrt{2}\mu \frac{\Delta\gamma}{\sigma_y}} \quad (22)$$

where $s_{n+1}^{\text{trial}} = s_n + 2\mu\Delta t\eta$. A final radial return adjustment is used to move the stress to the yield surface

$$s_{n+1} = \sqrt{\frac{2}{3}}\sigma_y \frac{\tilde{s}_{n+1}}{\|\tilde{s}_{n+1}\|} \quad (23)$$

A pathological situation arises if $\gamma_n = \mathbf{u}_n : \boldsymbol{\eta}_n$ is less than or equal to zero or $\Delta\gamma_{\text{er}} \geq \frac{d^*}{3}\Delta t$. This can occur if the rate of plastic deformation is small compared to the rate of elastic deformation or if the timestep size is too small (see [2]). In such situations, we use a locally implicit stress update that uses Newton iterations (as discussed in [6], page 124) to compute \tilde{s} .

3.2 Models

Below are some of the strain-rate, strain, and temperature dependent models for metals that are implemented in Uintah.

3.3 Equation of State Models

The elastic-plastic stress update assumes that the volumetric part of the Cauchy stress can be calculated using an equation of state. There are three equations of state that are implemented in Uintah. These are

1. A default hypoelastic equation of state.
2. A neo-Hookean equation of state.
3. A Mie-Gruneisen type equation of state.

Default hypoelastic equation of state In this case we assume that the stress rate is given by

$$\dot{\boldsymbol{\sigma}} = \lambda \text{tr}(\mathbf{d}^e) \mathbf{1} + 2\mu \mathbf{d}^e \quad (24)$$

where $\boldsymbol{\sigma}$ is the Cauchy stress, \mathbf{d}^e is the elastic part of the rate of deformation, and λ, μ are constants.

If $\boldsymbol{\eta}^e$ is the deviatoric part of \boldsymbol{d}^e then we can write

$$\dot{\boldsymbol{\sigma}} = \left(\lambda + \frac{2}{3} \mu \right) \text{tr}(\boldsymbol{d}^e) \mathbf{1} + 2 \mu \boldsymbol{\eta}^e = \kappa \text{tr}(\boldsymbol{d}^e) \mathbf{1} + 2 \mu \boldsymbol{\eta}^e . \quad (25)$$

If we split $\boldsymbol{\sigma}$ into a volumetric and a deviatoric part, i.e., $\boldsymbol{\sigma} = p \mathbf{1} + \boldsymbol{s}$ and take the time derivative to get $\dot{\boldsymbol{\sigma}} = \dot{p} \mathbf{1} + \dot{\boldsymbol{s}}$ then

$$\dot{p} = \kappa \text{tr}(\boldsymbol{d}^e) . \quad (26)$$

In addition we assume that $\boldsymbol{d} = \boldsymbol{d}^e + \boldsymbol{d}^p$. If we also assume that the plastic volume change is negligible, we can then write that

$$\dot{p} = \kappa \text{tr}(\boldsymbol{d}) . \quad (27)$$

This is the equation that is used to calculate the pressure p in the default hypoelastic equation of state, i.e.,

$$\boxed{p_{n+1} = p_n + \kappa \text{tr}(\boldsymbol{d}_{n+1}) \Delta t .} \quad (28)$$

To get the derivative of p with respect to J , where $J = \det(\boldsymbol{F})$, we note that

$$\dot{p} = \frac{\partial p}{\partial J} \dot{J} = \frac{\partial p}{\partial J} J \text{tr}(\boldsymbol{d}) . \quad (29)$$

Therefore,

$$\boxed{\frac{\partial p}{\partial J} = \frac{\kappa}{J} .} \quad (30)$$

This model is invoked in Uintah using

```
<equation_of_state type="default_hypo">
</equation_of_state>
```

The code is in

```
.../MPM/ConstitutiveModel/PlasticityModels/DefaultHypoElasticEOS.cc.
```

Default hyperelastic equation of state In this model the pressure is computed using the relation

$$p = \frac{1}{2} \kappa \left(J^e - \frac{1}{J^e} \right) \quad (31)$$

where κ is the bulk modulus and J^e is determinant of the elastic part of the deformation gradient.

We can also compute

$$\frac{dp}{dJ} = \frac{1}{2} \kappa \left(1 + \frac{1}{(J^e)^2} \right) . \quad (32)$$

This model is invoked in Uintah using

```
<equation_of_state type="default_hyper">
</equation_of_state>
```

The code is in `.../MPM/ConstitutiveModel/PlasticityModels/HyperElasticEOS.cc`. If an EOS is not specified then this model is the **default**.

Mie-Grüneisen equation of state The pressure (p) is calculated using a Mie-Grüneisen equation of state of the form ([7, 5])

$$p_{n+1} = -\frac{\rho_0 C_0^2 (1 - J_{n+1}^e)[1 - \Gamma_0(1 - J_{n+1}^e)/2]}{[1 - S_\alpha(1 - J_{n+1}^e)]^2} - \Gamma_0 e_{n+1}; \quad J^e := \det \mathbf{F}^e \quad (33)$$

where C_0 is the bulk speed of sound, ρ_0 is the initial mass density, $2\Gamma_0$ is the Grüneisen's gamma at the reference state, $S_\alpha = dU_s/dU_p$ is a linear Hugoniot slope coefficient, U_s is the shock wave velocity, U_p is the particle velocity, and e is the internal energy density (per unit reference volume), \mathbf{F}^e is the elastic part of the deformation gradient. For isochoric plasticity,

$$J^e = J = \det(\mathbf{F}) = \frac{\rho_0}{\rho}.$$

The internal energy is computed using

$$E = \frac{1}{V_0} \int C_v dT \approx \frac{C_v(T - T_0)}{V_0} \quad (34)$$

where $V_0 = 1/\rho_0$ is the reference specific volume at temperature $T = T_0$, and C_v is the specific heat at constant volume.

Also,

$$\frac{\partial p}{\partial J^e} = \frac{\rho_0 C_0^2 [1 + (S_\alpha - \Gamma_0)(1 - J^e)]}{[1 - S_\alpha(1 - J^e)]^3} - \Gamma_0 \frac{\partial e}{\partial J^e}. \quad (35)$$

We neglect the $\frac{\partial e}{\partial J^e}$ term in our calculations.

This model is invoked in Uintah using

```
<equation_of_state type="mie_gruneisen">
  <C_0>5386</C_0>
  <Gamma_0>1.99</Gamma_0>
  <S_alpha>1.339</S_alpha>
</equation_of_state>
```

The code is in .../MPM/ConstitutiveModel/PlasticityModels/MieGruneisenEOS.cc.

3.3.1 Melting Temperature

Default model The default model is to use a constant melting temperature. This model is invoked using

```
<melting_temp_model type="constant_Tm">
</melting_temp_model>
```

SCG melt model We use a pressure dependent relation to determine the melting temperature (T_m). The Steinberg-Cochran-Guinan (SCG) melt model ([8]) has been used for our simulations of copper. This model is based on a modified Lindemann law and has the form

$$T_m(\rho) = T_{m0} \exp \left[2a \left(1 - \frac{1}{\eta} \right) \right] \eta^{2(\Gamma_0 - a - 1/3)}; \quad \eta = \frac{\rho}{\rho_0} \quad (36)$$

where T_{m0} is the melt temperature at $\eta = 1$, a is the coefficient of the first order volume correction to Grüneisen's gamma (Γ_0).

This model is invoked with

```
<melting_temp_model type="scg_Tm">
  <T_m0> 2310.0 </T_m0>
  <Gamma_0> 3.0 </Gamma_0>
  <a> 1.67 </a>
</melting_temp_model>
```

BPS melt model An alternative melting relation that is based on dislocation-mediated phase transitions - the Burakovsky-Preston-Silbar (BPS) model ([9]) can also be used. This model has been used to determine the melt temperature for 4340 steel. The BPS model has the form

$$T_m(p) = T_m(0) \left[\frac{1}{\eta} + \frac{1}{\eta^{4/3}} \frac{\mu'_0}{\mu_0} p \right]; \quad \eta = \left(1 + \frac{K'_0}{K_0} p \right)^{1/K'_0} \quad (37)$$

$$T_m(0) = \frac{\kappa \lambda \mu_0 v_{WS}}{8\pi \ln(z-1) k_b} \ln \left(\frac{\alpha^2}{4 b^2 \rho_c(T_m)} \right) \quad (38)$$

where p is the pressure, $\eta = \rho/\rho_0$ is the compression, μ_0 is the shear modulus at room temperature and zero pressure, $\mu'_0 = \partial\mu/\partial p$ is the derivative of the shear modulus at zero pressure, K_0 is the bulk modulus at room temperature and zero pressure, $K'_0 = \partial K/\partial p$ is the derivative of the bulk modulus at zero pressure, κ is a constant, $\lambda = b^3/v_{WS}$ where b is the magnitude of the Burgers' vector, v_{WS} is the Wigner-Seitz volume, z is the coordination number, α is a constant, $\rho_c(T_m)$ is the critical density of dislocations, and k_b is the Boltzmann constant.

This model is invoked with

```
<melting_temp_model type="bps_Tm">
  <B0> 137e9 </B0>
  <dB_dp0> 5.48 </dB_dp0>
  <G0> 47.7e9 </G0>
  <dG_dp0> 1.4 </dG_dp0>
  <kappa> 1.25 </kappa>
  <z> 12 </z>
  <b2rhoTm> 0.64 </b2rhoTm>
  <alpha> 2.9 </alpha>
  <lambda> 1.41 </lambda>
```

```

<a> 3.6147e-9<a>
<v_ws_a3_factor> 1/4 <v_ws_a3_factor>
<Boltzmann_Constant> <Boltzmann_Constant>
</melting_temp_model>

```

3.3.2 Shear Modulus

Three models for the shear modulus (μ) have been tested in our simulations. The first has been associated with the Mechanical Threshold Stress (MTS) model and we call it the MTS shear model. The second is the model used by Steinberg-Cochran-Guinan and we call it the SCG shear model while the third is a model developed by Nadal and Le Poac that we call the NP shear model.

Default model The default model gives a constant shear modulus. The model is invoked using

```

<shear_modulus_model type="constant_shear">
</shear_modulus_model>

```

MTS Shear Modulus Model The simplest model is of the form suggested by [10] ([11])

$$\mu(T) = \mu_0 - \frac{D}{\exp(T_0/T) - 1} \quad (39)$$

where μ_0 is the shear modulus at 0K, and D, T_0 are material constants.

The model is invoked using

```

<shear_modulus_model type="mts_shear">
  <mu_0>28.0e9</mu_0>
  <D>4.50e9</D>
  <T_0>294</T_0>
</shear_modulus_model>

```

SCG Shear Modulus Model The Steinberg-Cochran-Guinan (SCG) shear modulus model ([8, 5]) is pressure dependent and has the form

$$\mu(p, T) = \mu_0 + \frac{\partial \mu}{\partial p} \frac{p}{\eta^{1/3}} + \frac{\partial \mu}{\partial T} (T - 300); \quad \eta = \rho/\rho_0 \quad (40)$$

where, μ_0 is the shear modulus at the reference state ($T = 300$ K, $p = 0$, $\eta = 1$), p is the pressure, and T is the temperature. When the temperature is above T_m , the shear modulus is instantaneously set to zero in this model.

The model is invoked using

```

<shear_modulus_model type="scg_shear">
  <mu_0> 81.8e9 </mu_0>
  <A> 20.6e-12 </A>
  <B> 0.16e-3 </B>
</shear_modulus_model>

```

NP Shear Modulus Model A modified version of the SCG model has been developed by [12] that attempts to capture the sudden drop in the shear modulus close to the melting temperature in a smooth manner. The Nadal-LePoac (NP) shear modulus model has the form

$$\mu(p, T) = \frac{1}{\mathcal{J}(\hat{T})} \left[\left(\mu_0 + \frac{\partial \mu}{\partial p} \frac{p}{\eta^{1/3}} \right) (1 - \hat{T}) + \frac{\rho}{Cm} k_b T \right]; \quad C := \frac{(6\pi^2)^{2/3}}{3} f^2 \quad (41)$$

where

$$\mathcal{J}(\hat{T}) := 1 + \exp \left[-\frac{1 + 1/\zeta}{1 + \zeta/(1 - \hat{T})} \right] \quad \text{for} \quad \hat{T} := \frac{T}{T_m} \in [0, 1 + \zeta], \quad (42)$$

μ_0 is the shear modulus at 0 K and ambient pressure, ζ is a material parameter, k_b is the Boltzmann constant, m is the atomic mass, and f is the Lindemann constant.

The model is invoked using

```

<shear_modulus_model type="np_shear">
  <mu_0>26.5e9</mu_0>
  <zeta>0.04</zeta>
  <slope_mu_p_over_mu0>65.0e-12</slope_mu_p_over_mu0>
  <C> 0.047 </C>
  <m> 26.98 </m>
</shear_modulus_model>

```

PTW Shear model The PTW shear model is a simplified version of the SCG shear model. The inputs can be found in `.../MPM/ConstitutiveModel/PlasticityModel/PTWShear.h`.

3.3.3 Flow Stress

We have explored five temperature and strain rate dependent models that can be used to compute the flow stress:

1. the Johnson-Cook (JC) model
2. the Steinberg-Cochran-Guinan-Lund (SCG) model.
3. the Zerilli-Armstrong (ZA) model.
4. the Mechanical Threshold Stress (MTS) model.
5. the Preston-Tonks-Wallace (PTW) model.

JC Flow Stress Model The Johnson-Cook (JC) model ([13]) is purely empirical and gives the following relation for the flow stress (σ_y)

$$\sigma_y(\varepsilon_p, \dot{\varepsilon}_p, T) = [A + B(\varepsilon_p)^n] [1 + C \ln(\dot{\varepsilon}_p^*)] [1 - (T^*)^m] \quad (43)$$

where ε_p is the equivalent plastic strain, $\dot{\varepsilon}_p$ is the plastic strain rate, A, B, C, n, m are material constants,

$$\dot{\varepsilon}_p^* = \frac{\dot{\varepsilon}_p}{\dot{\varepsilon}_{p0}}; \quad T^* = \frac{(T - T_0)}{(T_m - T_0)}, \quad (44)$$

$\dot{\varepsilon}_{p0}$ is a user defined plastic strain rate, T_0 is a reference temperature, and T_m is the melt temperature. For conditions where $T^* < 0$, we assume that $m = 1$.

The inputs for this model are

```
<plasticity_model type="johnson_cook">
  <A>792.0e6</A>
  <B>510.0e6</B>
  <C>0.014</C>
  <n>0.26</n>
  <m>1.03</m>
  <T_r>298.0</T_r>
  <T_m>1793.0</T_m>
  <epdot_0>1.0</epdot_0>
</plasticity_model>
```

SCG Flow Stress Model The Steinberg-Cochran-Guinan-Lund (SCG) model is a semi-empirical model that was developed by [8] for high strain rate situations and extended to low strain rates and bcc materials by [14]. The flow stress in this model is given by

$$\sigma_y(\varepsilon_p, \dot{\varepsilon}_p, T) = [\sigma_a f(\varepsilon_p) + \sigma_t(\dot{\varepsilon}_p, T)] \frac{\mu(p, T)}{\mu_0} \quad (45)$$

where σ_a is the athermal component of the flow stress, $f(\varepsilon_p)$ is a function that represents strain hardening, σ_t is the thermally activated component of the flow stress, $\mu(p, T)$ is the shear modulus, and μ_0 is the shear modulus at standard temperature and pressure. The strain hardening function has the form

$$f(\varepsilon_p) = [1 + \beta(\varepsilon_p + \varepsilon_{pi})]^n; \quad \sigma_a f(\varepsilon_p) \leq \sigma_{\max} \quad (46)$$

where β, n are work hardening parameters, and ε_{pi} is the initial equivalent plastic strain. The thermal component σ_t is computed using a bisection algorithm from the following equation (based on the work of [15])

$$\dot{\varepsilon}_p = \left[\frac{1}{C_1} \exp \left[\frac{2U_k}{k_b T} \left(1 - \frac{\sigma_t}{\sigma_p} \right)^2 \right] + \frac{C_2}{\sigma_t} \right]^{-1}; \quad \sigma_t \leq \sigma_p \quad (47)$$

where $2U_k$ is the energy to form a kink-pair in a dislocation segment of length L_d , k_b is the Boltzmann constant, σ_p is the Peierls stress. The constants C_1, C_2 are given by the relations

$$C_1 := \frac{\rho_d L_d a b^2 \nu}{2w^2}; \quad C_2 := \frac{D}{\rho_d b^2} \quad (48)$$

where ρ_d is the dislocation density, L_d is the length of a dislocation segment, a is the distance between Peierls valleys, b is the magnitude of the Burgers' vector, ν is the Debye frequency, w is the width of a kink loop, and D is the drag coefficient.

The inputs for this model are of the form

```
<plasticity_model type="steinberg_cochran_guinan">
  <mu_0> 81.8e9 </mu_0>
  <sigma_0> 1.15e9 </sigma_0>
  <Y_max> 0.25e9 </Y_max>
  <beta> 2.0 </beta>
  <n> 0.50 </n>
  <A> 20.6e-12 </A>
  <B> 0.16e-3 </B>
  <T_m0> 2310.0 </T_m0>
  <Gamma_0> 3.0 </Gamma_0>
  <a> 1.67 </a>
  <epsilon_p0> 0.0 </epsilon_p0>
</plasticity_model>
```

ZA Flow Stress Model The Zerilli-Armstrong (ZA) model ([16, 17, 18]) is based on simplified dislocation mechanics. The general form of the equation for the flow stress is

$$\sigma_y(\varepsilon_p, \dot{\varepsilon}_p, T) = \sigma_a + B \exp(-\beta(\dot{\varepsilon}_p)T) + B_0 \sqrt{\varepsilon_p} \exp(-\alpha(\dot{\varepsilon}_p)T) \quad (49)$$

where σ_a is the athermal component of the flow stress given by

$$\sigma_a := \sigma_g + \frac{k_h}{\sqrt{l}} + K \varepsilon_p^n, \quad (50)$$

σ_g is the contribution due to solutes and initial dislocation density, k_h is the microstructural stress intensity, l is the average grain diameter, K is zero for fcc materials, B, B_0 are material constants. The functional forms of the exponents α and β are

$$\alpha = \alpha_0 - \alpha_1 \ln(\dot{\varepsilon}_p); \quad \beta = \beta_0 - \beta_1 \ln(\dot{\varepsilon}_p); \quad (51)$$

where $\alpha_0, \alpha_1, \beta_0, \beta_1$ are material parameters that depend on the type of material (fcc, bcc, hcp, alloys). The Zerilli-Armstrong model has been modified by [19] for better performance at high temperatures. However, we have not used the modified equations in our computations.

The inputs for this model are of the form

```
<bcc_or_fcc> fcc </bcc_or_fcc>
<c2> </c2>
<c3> </c3>
<c4> </c4>
<n> </n>
```

MTS Flow Stress Model The Mechanical Threshold Stress (MTS) model ([20, 21, 22]) gives the following form for the flow stress

$$\sigma_y(\varepsilon_p, \dot{\varepsilon}_p, T) = \sigma_a + (S_i \sigma_i + S_e \sigma_e) \frac{\mu(p, T)}{\mu_0} \quad (52)$$

where σ_a is the athermal component of mechanical threshold stress, μ_0 is the shear modulus at 0 K and ambient pressure, σ_i is the component of the flow stress due to intrinsic barriers to thermally activated dislocation motion and dislocation-dislocation interactions, σ_e is the component of the flow stress due to microstructural evolution with increasing deformation (strain hardening), (S_i, S_e) are temperature and strain rate dependent scaling factors. The scaling factors take the Arrhenius form

$$S_i = \left[1 - \left(\frac{k_b T}{g_{0i} b^3 \mu(p, T)} \ln \frac{\dot{\varepsilon}_{p0i}}{\dot{\varepsilon}_p} \right)^{1/q_i} \right]^{1/p_i} \quad (53)$$

$$S_e = \left[1 - \left(\frac{k_b T}{g_{0e} b^3 \mu(p, T)} \ln \frac{\dot{\varepsilon}_{p0e}}{\dot{\varepsilon}_p} \right)^{1/q_e} \right]^{1/p_e} \quad (54)$$

where k_b is the Boltzmann constant, b is the magnitude of the Burgers' vector, (g_{0i}, g_{0e}) are normalized activation energies, $(\dot{\varepsilon}_{p0i}, \dot{\varepsilon}_{p0e})$ are constant reference strain rates, and (q_i, p_i, q_e, p_e) are constants. The strain hardening component of the mechanical threshold stress (σ_e) is given by a modified Voce law

$$\frac{d\sigma_e}{d\varepsilon_p} = \theta(\sigma_e) \quad (55)$$

where

$$\theta(\sigma_e) = \theta_0 [1 - F(\sigma_e)] + \theta_{IV} F(\sigma_e) \quad (56)$$

$$\theta_0 = a_0 + a_1 \ln \dot{\varepsilon}_p + a_2 \sqrt{\dot{\varepsilon}_p} - a_3 T \quad (57)$$

$$F(\sigma_e) = \frac{\tanh \left(\alpha \frac{\sigma_e}{\sigma_{es}} \right)}{\tanh(\alpha)} \quad (58)$$

$$\ln \left(\frac{\sigma_{es}}{\sigma_{0es}} \right) = \left(\frac{kT}{g_{0es} b^3 \mu(p, T)} \right) \ln \left(\frac{\dot{\varepsilon}_p}{\dot{\varepsilon}_{p0es}} \right) \quad (59)$$

and θ_0 is the hardening due to dislocation accumulation, θ_{IV} is the contribution due to stage-IV hardening, $(a_0, a_1, a_2, a_3, \alpha)$ are constants, σ_{es} is the stress at zero strain hardening rate, σ_{0es} is the saturation threshold stress for deformation at 0 K, g_{0es} is a constant, and $\dot{\varepsilon}_{p0es}$ is the maximum strain rate. Note that the maximum strain rate is usually limited to about 10^7 /s.

The inputs for this model are of the form

```
<plasticity_model type="mts_model">
  <sigma_a>363.7e6</sigma_a>
  <mu_0>28.0e9</mu_0>
  <D>4.50e9</D>
  <T_0>294</T_0>
  <koverbcubed>0.823e6</koverbcubed>
```

```

<g_0i>0.0</g_0i>
<g_0e>0.71</g_0e>
<edot_0i>0.0</edot_0i>
<edot_0e>2.79e9</edot_0e>
<p_i>0.0</p_i>
<q_i>0.0</q_i>
<p_e>1.0</p_e>
<q_e>2.0</q_e>
<sigma_i>0.0</sigma_i>
<a_0>211.8e6</a_0>
<a_1>0.0</a_1>
<a_2>0.0</a_2>
<a_3>0.0</a_3>
<theta_IV>0.0</theta_IV>
<alpha>2</alpha>
<edot_es0>3.42e8</edot_es0>
<g_0es>0.15</g_0es>
<sigma_es0>1679.3e6</sigma_es0>
</plasticity_model>

```

PTW Flow Stress Model The Preston-Tonks-Wallace (PTW) model ([23]) attempts to provide a model for the flow stress for extreme strain rates (up to 10^{11} /s) and temperatures up to melt. The flow stress is given by

$$\sigma_y(\varepsilon_p, \dot{\varepsilon}_p, T) = \begin{cases} 2 \left[\tau_s + \alpha \ln \left[1 - \varphi \exp \left(-\beta - \frac{\theta \varepsilon_p}{\alpha \varphi} \right) \right] \right] \mu(p, T) & \text{thermal regime} \\ 2\tau_s \mu(p, T) & \text{shock regime} \end{cases} \quad (60)$$

with

$$\alpha := \frac{s_0 - \tau_y}{d}; \quad \beta := \frac{\tau_s - \tau_y}{\alpha}; \quad \varphi := \exp(\beta) - 1 \quad (61)$$

where τ_s is a normalized work-hardening saturation stress, s_0 is the value of τ_s at 0K, τ_y is a normalized yield stress, θ is the hardening constant in the Voce hardening law, and d is a dimensionless material parameter that modifies the Voce hardening law. The saturation stress and the yield stress are given by

$$\tau_s = \max \left\{ s_0 - (s_0 - s_\infty) \operatorname{erf} \left[\kappa \hat{T} \ln \left(\frac{\gamma \dot{\xi}}{\dot{\varepsilon}_p} \right) \right], s_0 \left(\frac{\dot{\varepsilon}_p}{\gamma \dot{\xi}} \right)^{s_1} \right\} \quad (62)$$

$$\tau_y = \max \left\{ y_0 - (y_0 - y_\infty) \operatorname{erf} \left[\kappa \hat{T} \ln \left(\frac{\gamma \dot{\xi}}{\dot{\varepsilon}_p} \right) \right], \min \left\{ y_1 \left(\frac{\dot{\varepsilon}_p}{\gamma \dot{\xi}} \right)^{y_2}, s_0 \left(\frac{\dot{\varepsilon}_p}{\gamma \dot{\xi}} \right)^{s_1} \right\} \right\} \quad (63)$$

where s_∞ is the value of τ_s close to the melt temperature, (y_0, y_∞) are the values of τ_y at 0K and close to melt, respectively, (κ, γ) are material constants, $\hat{T} = T/T_m$, (s_1, y_1, y_2) are material parameters for the high strain rate regime, and

$$\dot{\xi} = \frac{1}{2} \left(\frac{4\pi\rho}{3M} \right)^{1/3} \left(\frac{\mu(p, T)}{\rho} \right)^{1/2} \quad (64)$$

where ρ is the density, and M is the atomic mass.

The inputs for this model are of the form

```
<plasticity_model type="preston_tonks_wallace">
  <theta> 0.025 </theta>
  <p> 2.0 </p>
  <s0> 0.0085 </s0>
  <sinf> 0.00055 </sinf>
  <kappa> 0.11 </kappa>
  <gamma> 0.00001 </gamma>
  <y0> 0.0001 </y0>
  <yinf> 0.0001 </yinf>
  <y1> 0.094 </y1>
  <y2> 0.575 </y2>
  <beta> 0.25 </beta>
  <M> 63.54 </M>
  <G0> 518e8 </G0>
  <alpha> 0.20 </alpha>
  <alphap> 0.20 </alphap>
</plasticity_model>
```

3.3.4 Adiabatic Heating and Specific Heat

A part of the plastic work done is converted into heat and used to update the temperature of a particle. The increase in temperature (ΔT) due to an increment in plastic strain ($\Delta \epsilon_p$) is given by the equation

$$\Delta T = \frac{\chi \sigma_y}{\rho C_p} \Delta \epsilon_p \quad (65)$$

where χ is the Taylor-Quinney coefficient, and C_p is the specific heat. The value of the Taylor-Quinney coefficient is taken to be 0.9 in all our simulations (see [24] for more details on the variation of χ with strain and strain rate).

The Taylor-Quinney coefficient is taken as input in the ElasticPlastic model using the tags

```
<taylor_quinney_coeff> 0.9 </taylor_quinney_coeff>
```

Default specific heat model The default model returns a constant specific heat and is invoked using

```
<specific_heat_model type="constant_Cp">
</specific_heat_model>
```

Specific heat model for copper The specific heat model for copper is of the form

$$C_p = \begin{cases} A_0 T^3 - B_0 T^2 + C_0 T - D_0 & \text{if } T < T_0 \\ A_1 T + B_1 & \text{if } T \geq T_0 . \end{cases} \quad (66)$$

The model is invoked using

```
<specific_heat_model type = "copper_Cp"> </specific_heat_model>
```

Specific heat model for steel A relation for the dependence of C_p upon temperature is used for the steel ([25]).

$$C_p = \begin{cases} A_1 + B_1 t + C_1 |t|^{-\alpha} & \text{if } T < T_c \\ A_2 + B_2 t + C_2 t^{-\alpha'} & \text{if } T > T_c \end{cases} \quad (67)$$

$$t = \frac{T}{T_c} - 1 \quad (68)$$

where T_c is the critical temperature at which the phase transformation from the α to the γ phase takes place, and $A_1, A_2, B_1, B_2, \alpha, \alpha'$ are constants.

The model is invoked using

```
<specific_heat_model type = "steel_Cp"> </specific_heat_model>
```

The heat generated at a material point is conducted away at the end of a time step using the transient heat equation. The effect of conduction on material point temperature is negligible (but non-zero) for the high strain-rate problems simulated using Uintah.

3.4 Adding new models

In the parallel implementation of the stress update algorithm, sockets have been added to allow for the incorporation of a variety of plasticity, damage, yield, and bifurcation models without requiring any change in the stress update code. The algorithm is shown in Algorithm 1. The equation of state, plasticity model, yield condition, damage model, and the stability criterion are all polymorphic objects created using a factory idiom in C++ ([26]).

Addition of a new model requires the following steps (the example below is only for the flow stress model but the same idea applies to other models) :

1. Creation of a new class that encapsulates the plasticity model. The template for this class can be copied from the existing plasticity models. The data that is unique to the new model are specified in the form of
 - A structure containing the constants for the plasticity model.

Table 1: Stress Update Algorithm

Persistent: Initial moduli, temperature, porosity,
 scalar damage, equation of state, plasticity model,
 yield condition, stability criterion, damage model

Temporary: Particle state at time t

Output: Particle state at time $t + \Delta t$

For all the patches in the domain
 Read the particle data and initialize updated data storage
For all the particles in the patch
 Compute the velocity gradient and the rate of deformation tensor
 Compute the deformation gradient and the rotation tensor
 Rotate the Cauchy stress and the rate of deformation tensor
 to the material configuration
 Compute the current shear modulus and melting temperature
 Compute the pressure using the equation of state,
 update the hydrostatic stress, and
 compute the trial deviatoric stress
 Compute the flow stress using the plasticity model
 Evaluate the yield function
If *particle is elastic*
 Update the elastic deviatoric stress from the trial stress
 Rotate the stress back to laboratory coordinates
 Update the particle state
Else
 Compute the elastic-plastic deviatoric stress
 Compute updated porosity, scalar damage, and
 temperature increase due to plastic work
 Compute elastic-plastic tangent modulus and evaluate stability condition
 Rotate the stress back to laboratory coordinates
 Update the particle state
End If
If *Temperature > Melt Temperature or Porosity > Critical Porosity or Unstable*
 Tag particle as failed
End If
 Convert failed particles into a material with a different velocity field
End For
End For

- Particle variables that specify the variables that evolve in the plasticity model.
2. The implementation of the plasticity model involves the following steps.
 - Reading the input file for the model constants in the constructor.
 - Adding the variables that evolve in the plasticity model appropriately to the task graph.
 - Adding the appropriate flow stress calculation method.
 3. The `PlasticityModelFactory` is then modified so that it recognizes the added plasticity model.

3.5 Damage Models and Failure

Only the Johnson-Cook damage evolution rule has been added to the `DamageModelFactory` so far. The damage model framework is designed to be similar to the plasticity model framework. New models can be added using the approach described in the previous section.

A particle is tagged as “failed” when its temperature is greater than the melting point of the material at the applied pressure. An additional condition for failure is when the porosity of a particle increases beyond a critical limit and the strain exceeds the fracture strain of the material. Another condition for failure is when a material bifurcation condition such as the Drucker stability postulate is satisfied. Upon failure, a particle is either removed from the computation by setting the stress to zero or is converted into a material with a different velocity field which interacts with the remaining particles via contact. Either approach leads to the simulation of a newly created surface. More details of the approach can be found in [27, 28, 29].

3.5.1 Yield conditions

When failure is to be simulated we can use the Gurson-Tvergaard-Needleman yield condition instead of the von Mises condition.

The von Mises yield condition The von Mises yield condition is the default and is invoked using the tags

```
<yield_condition type="vonMises">
</yield_condition>
```

The Gurson-Tvergaard-Needleman (GTN) yield condition The Gurson-Tvergaard-Needleman (GTN) yield condition [30, 31] depends on porosity. An associated flow rule is used to determine the plastic rate parameter in either case. The GTN yield condition can be written as

$$\Phi = \left(\frac{\sigma_{eq}}{\sigma_f} \right)^2 + 2q_1 f_* \cosh \left(q_2 \frac{Tr(\sigma)}{2\sigma_f} \right) - (1 + q_3 f_*^2) = 0 \quad (69)$$

where q_1, q_2, q_3 are material constants and f_* is the porosity (damage) function given by

$$f_* = \begin{cases} f & \text{for } f \leq f_c, \\ f_c + k(f - f_c) & \text{for } f > f_c \end{cases} \quad (70)$$

where k is a constant and f is the porosity (void volume fraction). The flow stress in the matrix material is computed using either of the two plasticity models discussed earlier. Note that the flow stress in the matrix material also remains on the undamaged matrix yield surface and uses an associated flow rule.

This yield condition is invoked using

```
<yield_condition type="gurson">
  <q1> 1.5 </q1>
  <q2> 1.0 </q2>
  <q3> 2.25 </q3>
  <k> 4.0 </k>
  <f_c> 0.05 </f_c>
</yield_condition>
```

3.5.2 Porosity model

The evolution of porosity is calculated as the sum of the rate of growth and the rate of nucleation [32]. The rate of growth of porosity and the void nucleation rate are given by the following equations [33]

$$\dot{f} = \dot{f}_{\text{nucl}} + \dot{f}_{\text{grow}} \quad (71)$$

$$\dot{f}_{\text{grow}} = (1 - f) \text{Tr}(\mathbf{D}_p) \quad (72)$$

$$\dot{f}_{\text{nucl}} = \frac{f_n}{(s_n \sqrt{2\pi})} \exp \left[-\frac{1}{2} \frac{(\epsilon_p - \epsilon_n)^2}{s_n^2} \right] \dot{\epsilon}_p \quad (73)$$

where \mathbf{D}_p is the rate of plastic deformation tensor, f_n is the volume fraction of void nucleating particles, ϵ_n is the mean of the distribution of nucleation strains, and s_n is the standard deviation of the distribution.

The inputs tags for porosity are of the form

```
<evolve_porosity> true </evolve_porosity>
<initial_mean_porosity> 0.005 </initial_mean_porosity>
<initial_std_porosity> 0.001 </initial_std_porosity>
<critical_porosity> 0.3 </critical_porosity>
<frac_nucleation> 0.1 </frac_nucleation>
<meanstrain_nucleation> 0.3 </meanstrain_nucleation>
<stddevstrain_nucleation> 0.1 </stddevstrain_nucleation>
<initial_porosity_distrib> gauss </initial_porosity_distrib>
```

3.5.3 Damage model

After the stress state has been determined on the basis of the yield condition and the associated flow rule, a scalar damage state in each material point can be calculated using the Johnson-Cook model [34]. The Johnson-Cook model has an explicit dependence on temperature, plastic strain, and strain rate.

The damage evolution rule for the Johnson-Cook damage model can be written as

$$\dot{D} = \frac{\dot{\epsilon}_p}{\epsilon_p^f}; \quad \epsilon_p^f = \left[D_1 + D_2 \exp\left(\frac{D_3}{3} \sigma^*\right) \right] [1 + D_4 \ln(\dot{\epsilon}_p^*)] [1 + D_5 T^*]; \quad \sigma^* = \frac{\text{Tr}(\boldsymbol{\sigma})}{\sigma_{eq}}; \quad (74)$$

where D is the damage variable which has a value of 0 for virgin material and a value of 1 at fracture, ϵ_p^f is the fracture strain, D_1, D_2, D_3, D_4, D_5 are constants, $\boldsymbol{\sigma}$ is the Cauchy stress, and T^* is the scaled temperature as in the Johnson-Cook plasticity model.

The input tags for the damage model are :

```
<damage_model type="johnson_cook">
  <D1>0.05</D1>
  <D2>3.44</D2>
  <D3>-2.12</D3>
  <D4>0.002</D4>
  <D5>0.61</D5>
</damage_model>
```

An initial damage distribution can be created using the following tags

```
<evolve_damage> true </evolve_damage>
<initial_mean_scalar_damage> 0.005 </initial_mean_scalar_damage>
<initial_std_scalar_damage> 0.001 </initial_std_scalar_damage>
<critical_scalar_damage> 1.0 </critical_scalar_damage>
<initial_scalar_damage_distrib> gauss </initial_scalar_damage_distrib>
```

3.6 Erosion algorithm

Under normal conditions, the heat generated at a material point is conducted away at the end of a time step using the heat equation. If special adiabatic conditions apply (such as in impact problems), the heat is accumulated at a material point and is not conducted to the surrounding particles. This localized heating can be used to determine whether a material point has melted.

The determination of whether a particle has failed can be made on the basis of either or all of the following conditions:

- The particle temperature exceeds the melting temperature.

- The TEPLA-F fracture condition [35] is satisfied. This condition can be written as

$$(f/f_c)^2 + (\epsilon_p/\epsilon_p^f)^2 = 1 \quad (75)$$

where f is the current porosity, f_c is the maximum allowable porosity, ϵ_p is the current plastic strain, and ϵ_p^f is the plastic strain at fracture.

- An alternative to ad-hoc damage criteria is to use the concept of bifurcation to determine whether a particle has failed or not. Two stability criteria have been explored in this paper - the Drucker stability postulate [36] and the loss of hyperbolicity criterion (using the determinant of the acoustic tensor) [37, 38].

The simplest criterion that can be used is the Drucker stability postulate [36] which states that time rate of change of the rate of work done by a material cannot be negative. Therefore, the material is assumed to become unstable (and a particle fails) when

$$\dot{\sigma} : D^p \leq 0 \quad (76)$$

Another stability criterion that is less restrictive is the acoustic tensor criterion which states that the material loses stability if the determinant of the acoustic tensor changes sign [37, 38]. Determination of the acoustic tensor requires a search for a normal vector around the material point and is therefore computationally expensive. A simplification of this criterion is a check which assumes that the direction of instability lies in the plane of the maximum and minimum principal stress [39]. In this approach, we assume that the strain is localized in a band with normal \mathbf{n} , and the magnitude of the velocity difference across the band is \mathbf{g} . Then the bifurcation condition leads to the relation

$$R_{ij}g_j = 0; \quad R_{ij} = M_{ikjl}n_kn_l + M_{ilkj}n_kn_l - \sigma_{ik}n_jn_k \quad (77)$$

where M_{ijkl} are the components of the co-rotational tangent modulus tensor and σ_{ij} are the components of the co-rotational stress tensor. If $\det(R_{ij}) \leq 0$, then g_j can be arbitrary and there is a possibility of strain localization. If this condition for loss of hyperbolicity is met, then a particle deforms in an unstable manner and failure can be assumed to have occurred at that particle. We use a combination of these criteria to simulate failure.

Since the material in the container may unload locally after fracture, the hypoelastic-plastic stress update may not work accurately under certain circumstances. An improvement would be to use a hyperelastic-plastic stress update algorithm. Also, the plasticity models are temperature dependent. Hence there is the issue of severe mesh dependence due to change of the governing equations from hyperbolic to elliptic in the softening regime [40, 41, 42]. Viscoplastic stress update models or nonlocal/gradient plasticity models [43, 44] can be used to eliminate some of these effects and are currently under investigation.

The tags used to control the erosion algorithm are in two places. In the <MPM> </MPM> section the following flags can be set

```
<erosion algorithm = "ZeroStress"/>
<create_new_particles>           false      </create_new_particles>
<manual_new_material>           false      </manual_new_material>
```


Similarly, the damage model is called using a function of the type

```
double damage = d_damage->computeScalarDamage(tensorEta, tensorS,
                                              pTemperature[idx],
                                              delT, matl, d_tol,
                                              pDamage[idx]);
```

Therefore, the plasticity, damage and equation of state models are easily be inserted into any other type of stress update algorithm without any change being needed in them as can be seen in the hyperelastic-plastic stress update algorithm discussed below.

4 Example Input Files

4.1 Hypoelastic-plastic model

An example of the portion of an input file that specifies a copper body with a hypoelastic stress update, Johnson-Cook plasticity model, Johnson-Cook Damage Model and Mie-Gruneisen Equation of State is shown below.

```
<material>

<include href="inputs/MPM/MaterialData/MaterialConstAnnCopper.xml"/>
<constitutive_model type="hypoelastic_plastic">
  <tolerance>5.0e-10</tolerance>
  <include href="inputs/MPM/MaterialData/IsotropicElasticAnnCopper.xml"/>
  <include href="inputs/MPM/MaterialData/JohnsonCookPlasticAnnCopper.xml"/>
  <include href="inputs/MPM/MaterialData/JohnsonCookDamageAnnCopper.xml"/>
  <include href="inputs/MPM/MaterialData/MieGruneisenEOSAnnCopper.xml"/>
</constitutive_model>

<burn type = "null" />
<velocity_field>1</velocity_field>

<geom_object>
  <cylinder label = "Cylinder">
    <bottom>[0.0,0.0,0.0]</bottom>
    <top>[0.0,2.54e-2,0.0]</top>
    <radius>0.762e-2</radius>
  </cylinder>
  <res>[3,3,3]</res>
  <velocity>[0.0,-208.0,0.0]</velocity>
  <temperature>294</temperature>
</geom_object>

</material>
```

The general material constants for copper are in the file `MaterialConstAnnCopper.xml`. The contents are shown below

```
<?xml version='1.0' encoding='ISO-8859-1' ?>
<Uintah_Include>
  <density>8930.0</density>
  <toughness>10.e6</toughness>
  <thermal_conductivity>1.0</thermal_conductivity>
  <specific_heat>383</specific_heat>
  <room_temp>294.0</room_temp>
  <melt_temp>1356.0</melt_temp>
</Uintah_Include>
```

The elastic properties are in the file `IsotropicElasticAnnCopper.xml`. The contents of this file are shown below.

```
<?xml version='1.0' encoding='ISO-8859-1' ?>
<Uintah_Include>
  <shear_modulus>45.45e9</shear_modulus>
  <bulk_modulus>136.35e9</bulk_modulus>
</Uintah_Include>
```

The constants for the Johnson-Cook plasticity model are in the file `JohnsonCookPlasticAnnCopper.xml`. The contents of this file are shown below.

```
<?xml version='1.0' encoding='ISO-8859-1' ?>
<Uintah_Include>
  <plasticity_model type="johnson_cook">
    <A>89.6e6</A>
    <B>292.0e6</B>
    <C>0.025</C>
    <n>0.31</n>
    <m>1.09</m>
  </plasticity_model>
</Uintah_Include>
```

The constants for the Johnson-Cook damage model are in the file `JohnsonCookDamageAnnCopper.xml`. The contents of this file are shown below.

```
<?xml version='1.0' encoding='ISO-8859-1' ?>
<Uintah_Include>
  <damage_model type="johnson_cook">
    <D1>0.54</D1>
```

```

    <D2>4.89</D2>
    <D3>-3.03</D3>
    <D4>0.014</D4>
    <D5>1.12</D5>
  </damage_model>
</Uintah_Include>

```

The constants for the Mie-Gruneisen model (as implemented in the Uintah Computational Framework) are in the file `MieGruneisenEOSAnnCopper.xml`. The contents of this file are shown below.

```

<?xml version='1.0' encoding='ISO-8859-1' ?>
<Uintah_Include>
  <equation_of_state type="mie_gruneisen">
    <C_0>3940</C_0>
    <Gamma_0>2.02</Gamma_0>
    <S_alpha>1.489</S_alpha>
  </equation_of_state>
</Uintah_Include>

```

As can be seen from the input file, any other plasticity model, damage model and equation of state can be used to replace the Johnson-Cook and Mie-Gruneisen models without any extra effort (provided the models have been implemented and the data exist).

The material data can easily be taken from a material database or specified for a new material in an input file kept at a centralized location. At this stage material data for a range of materials is kept in the directory `.../Uintah/StandAlone/inputs/MPM/MaterialData`.

4.2 Elastic-plastic model

The `<constitutive_model type="elastic_plastic">` model is more stable (and also more general) than the `<constitutive_model type="hypoelastic_plastic">` model. A sample input file for this model is shown below.

```

<MPM>
  <do_grid_reset> false </do_grid_reset>
  <time_integrator>explicit</time_integrator>
  <boundary_traction_faces>[zminus,zplus]</boundary_traction_faces>
  <dynamic>true</dynamic>
  <solver>simple</solver>
  <convergence_criteria_disp>1.e-10</convergence_criteria_disp>
  <convergence_criteria_energy>4.e-10</convergence_criteria_energy>
  <DoImplicitHeatConduction>true</DoImplicitHeatConduction>
  <interpolator>linear</interpolator>
  <minimum_particle_mass> 1.0e-8</minimum_particle_mass>

```

```

<maximum_particle_velocity> 1.0e8</maximum_particle_velocity>
<artificial_damping_coeff> 0.0 </artificial_damping_coeff>
<artificial_viscosity> true </artificial_viscosity>
<accumulate_strain_energy> true </accumulate_strain_energy>
<use_load_curves> false </use_load_curves>
<turn_on_adiabatic_heating> false </turn_on_adiabatic_heating>
<do_contact_friction_heating> false </do_contact_friction_heating>
<create_new_particles> false </create_new_particles>
<erosion_algorithm = "none"/>
</MPM>

<MaterialProperties>
  <MPM>
    <material name = "OFHCCu">
      <density> 8930.0 </density>
      <thermal_conductivity> 386.0 </thermal_conductivity>
      <specific_heat> 414.0 </specific_heat>
      <room_temp> 294.0 </room_temp>
      <melt_temp> 1356.0 </melt_temp>
      <constitutive_model type="elastic_plastic">
        <isothermal> false </isothermal>
        <tolerance> 1.0e-12 </tolerance>
        <do_melting> false </do_melting>
        <evolve_porosity> false </evolve_porosity>
        <evolve_damage> false </evolve_damage>
        <check_TEPLA_failure_criterion> false </check_TEPLA_failure_criterion>
        <check_max_stress_failure> false </check_max_stress_failure>
        <initial_material_temperature> 696.0 </initial_material_temperature>

        <shear_modulus> 46.0e9 </shear_modulus>
        <bulk_modulus> 129.0e9 </bulk_modulus>
        <coeff_thermal_expansion> 1.76e-5 </coeff_thermal_expansion>
        <taylor_quinney_coeff> 0.9 </taylor_quinney_coeff>
        <critical_stress> 129.0e9 </critical_stress>

        <equation_of_state type = "mie_gruneisen">
          <C_0> 3940 </C_0>
          <Gamma_0> 2.02 </Gamma_0>
          <S_alpha> 1.489 </S_alpha>
        </equation_of_state>

        <plasticity_model type="mts_model">
          <sigma_a>40.0e6</sigma_a>
          <mu_0>47.7e9</mu_0>
          <D>3.0e9</D>
          <T_0>180</T_0>
          <koverbcubed>0.823e6</koverbcubed>

```

```

    <g_0i>0.0</g_0i>
    <g_0e>1.6</g_0e>
    <edot_0i>0.0</edot_0i>
    <edot_0e>1.0e7</edot_0e>
    <p_i>0.0</p_i>
    <q_i>0.0</q_i>
    <p_e>0.666667</p_e>
    <q_e>1.0</q_e>
    <sigma_i>0.0</sigma_i>
    <a_0>2390.0e6</a_0>
    <a_1>12.0e6</a_1>
    <a_2>1.696e6</a_2>
    <a_3>0.0</a_3>
    <theta_IV>0.0</theta_IV>
    <alpha>2</alpha>
    <edot_es0>1.0e7</edot_es0>
    <g_0es>0.2625</g_0es>
    <sigma_es0>770.0e6</sigma_es0>
</plasticity_model>

<shear_modulus_model type="mts_shear">
    <mu_0>47.7e9</mu_0>
    <D>3.0e9</D>
    <T_0>180</T_0>
</shear_modulus_model>

<melting_temp_model type = "constant_Tm">
</melting_temp_model>

<yield_condition type = "vonMises">
</yield_condition>

<stability_check type = "none">
</stability_check>

<damage_model type = "hancock_mackenzie">
    <D0> 0.0001 </D0>
    <Dc> 0.7 </Dc>
</damage_model>

<compute_specfic_heat> false </compute_specfic_heat>
<specific_heat_model type="constant_Cp">
</specific_heat_model>

</constitutive_model>
<geom_object>
    <box label = "box">

```

```

        <min>[0.0,    0.0,    0.0]</min>
        <max>[1.0e-2, 1.0e-2, 1.0e-2]</max>
    </box>
    <res>[1,1,1]</res>
    <velocity>[0.0, 0.0, 0.0]</velocity>
    <temperature>696</temperature>
</geom_object>
</material>

</MPM>
</MaterialProperties>

```

4.2.1 An exploding ring experiment

The following shows the complete input file for an expanding ring test.

```

<?xml version='1.0' encoding='ISO-8859-1' ?>
<Uintah_specification>
<!--Please use a consistent set of units, (mks, cgs,...)-->
    <!-- First crack at the tuna can problem -->

    <Meta>
        <title>Pressurization of a container via burning w/o fracture</title>
    </Meta>&gt;
    <SimulationComponent>
        <type> mpmic </type>
    </SimulationComponent>
    <!-- _____ -->
    <!--   T   I   M   E       V   A   R   I   A   B   L   E   S   -->
    <!-- _____ -->

    <Time>
        <max_Timesteps>    99999          </max_Timesteps>>
        <maxTime>          2.00e-2        </maxTime>
        <initTime>          0.0            </initTime>
        <delt_min>          1.0e-12        </delt_min>
        <delt_max>          1.0            </delt_max>
        <delt_init>         2.1e-8         </delt_init>
        <timestep_multiplier> 0.5          </timestep_multiplier>
    </Time>
    <!-- _____ -->
    <!--   G   R   I   D       V   A   R   I   A   B   L   E   S   -->
    <!-- _____ -->

    <Grid>
    <BoundaryConditions>
        <Face side = "x-">

```

```

    <BCType id = "all" label = "Symmetric" var = "symmetry">
    </BCType>
</Face>
<Face side = "x+">
    <BCType id = "0" label = "Pressure" var = "Neumann">
        <value> 0.0 </value>
    </BCType>
    <BCType id = "all" label = "Velocity" var = "Dirichlet">
        <value> [0.,0.,0.] </value>
    </BCType>
    <BCType id = "all" label = "Temperature" var = "Neumann">
        <value> 0.0 </value>
    </BCType>
    <BCType id = "all" label = "Density" var = "Neumann">
        <value> 0.0 </value>
    </BCType>
</Face>
<Face side = "y-">
    <BCType id = "all" label = "Symmetric" var = "symmetry">
    </BCType>
</Face>
<Face side = "y+">
    <BCType id = "0" label = "Pressure" var = "Neumann">
        <value> 0.0 </value>
    </BCType>
    <BCType id = "all" label = "Velocity" var = "Dirichlet">
        <value> [0.,0.,0.] </value>
    </BCType>
    <BCType id = "all" label = "Temperature" var = "Neumann">
        <value> 0.0 </value>
    </BCType>
    <BCType id = "all" label = "Density" var = "Neumann">
        <value> 0.0 </value>
    </BCType>
</Face>
<Face side = "z-">
    <BCType id = "all" label = "Symmetric" var = "symmetry">
    </BCType>
</Face>
<Face side = "z+">
    <BCType id = "all" label = "Symmetric" var = "symmetry">
    </BCType>
</Face>
</BoundaryConditions>
<Level>
    <Box label = "1">
        <lower>          [ -0.08636, -0.08636, -0.0016933] </lower>

```

```

        <upper>          [ 0.08636, 0.08636, 0.0016933]      </upper>
        <extraCells>    [1,1,1]      </extraCells>
        <patches>       [2,2,1]      </patches>
        <resolution>    [102, 102, 1]      </resolution>
    </Box>
</Level>
</Grid>

<!-- _____ -->
<!--   O U P U T       V A R I A B L E S   -->
<!-- _____ -->

<DataArchiver>
    <filebase>explodeRFull.uda</filebase>
    <outputTimestepInterval> 20 </outputTimestepInterval>
    <save label = "rho_CC"/>
    <save label = "press_CC"/>
    <save label = "temp_CC"/>
    <save label = "vol_frac_CC"/>
    <save label = "vel_CC"/>
    <save label = "g.mass"/>
    <save label = "p.x"/>
    <save label = "p.mass"/>
    <save label = "p.temperature"/>
    <save label = "p.porosity"/>
    <save label = "p.particleID"/>
    <save label = "p.velocity"/>
    <save label = "p.stress"/>
    <save label = "p.damage" material = "0"/>
    <save label = "p.plasticStrain" material = "0"/>
    <save label = "p.strainRate" material = "0"/>
    <save label = "g.stressFS"/>
    <save label = "delP_Dilatate"/>
    <save label = "delP_MassX"/>
    <save label = "p.localized"/>
    <checkpoint cycle = "2" timestepInterval = "20"/>
</DataArchiver>

<Debug>
</Debug>

<!-- _____ -->
<!--   I C E       P A R A M E T E R S   -->
<!-- _____ -->

<CFD>
    <cfl>0.5</cfl>
    <CanAddICEMaterial>true</CanAddICEMaterial>
    <ICE>
        <advection type = "SecondOrder"/>

```



```

    <ClampSpecificVolume>true</ClampSpecificVolume>
  </ICE>
</CFD>

<!-- _____ -->
<!--      P H Y S I C A L      C O N S T A N T S      -->
<!-- _____ -->
<PhysicalConstants>
  <gravity>          [0,0,0]    </gravity>
  <reference_pressure> 101325.0 </reference_pressure>
</PhysicalConstants>

<MPM>
  <time_integrator>      explicit    </time_integrator>
  <nodes8or27>           27          </nodes8or27>
  <minimum_particle_mass> 3.e-12     </minimum_particle_mass>
  <maximum_particle_velocity> 1.e3    </maximum_particle_velocity>
  <artificial_damping_coeff> 0.0      </artificial_damping_coeff>
  <artificial_viscosity>   true       </artificial_viscosity>
  <artificial_viscosity_coeff1> 0.07   </artificial_viscosity_coeff1>
  <artificial_viscosity_coeff2> 1.6     </artificial_viscosity_coeff2>
  <turn_on_adiabatic_heating> false    </turn_on_adiabatic_heating>
  <accumulate_strain_energy> false     </accumulate_strain_energy>
  <use_load_curves>        false       </use_load_curves>
  <create_new_particles>   false       </create_new_particles>
  <manual_new_material>    false       </manual_new_material>
  <DoThermalExpansion>     false       </DoThermalExpansion>
  <testForNegTemps_mpm>    false       </testForNegTemps_mpm>
  <erosion_algorithm = "ZeroStress"/>
</MPM>

<!-- _____ -->
<!--      MATERIAL PROPERTIES INITIAL CONDITIONS      -->
<!-- _____ -->
<MaterialProperties>
  <MPM>
    <material name = "Steel Ring">
      <include href="inputs/MPM/MaterialData/MatConst4340St.xml"/>
      <constitutive_model type="elastic_plastic">
        <isothermal>          false </isothermal>
        <tolerance>            1.0e-10 </tolerance>
        <evolve_porosity>      true  </evolve_porosity>
        <evolve_damage>        true  </evolve_damage>
        <compute_specific_heat> true  </compute_specific_heat>
        <do_melting>           true  </do_melting>
        <useModifiedEOS>       true  </useModifiedEOS>
        <check_TEPLA_failure_criterion> true </check_TEPLA_failure_criterion>
      </constitutive_model>
    </material>
  </MPM>
</MaterialProperties>

```

```

<initial_material_temperature> 600.0 </initial_material_temperature>
<taylor_quinney_coeff> 0.9 </taylor_quinney_coeff>
<check_max_stress_failure> false </check_max_stress_failure>
<critical_stress> 12.0e9 </critical_stress>

<!-- Warning: you must copy link this input file into your -->
<!-- sus directory or these paths won't work. -->

<include href="inputs/MPM/MaterialData/IsoElastic4340St.xml"/>
<include href="inputs/MPM/MaterialData/MieGrunEOS4340St.xml"/>
<include href="inputs/MPM/MaterialData/ConstantShear.xml"/>
<include href="inputs/MPM/MaterialData/ConstantTm.xml"/>
<include href="inputs/MPM/MaterialData/JCPlastic4340St.xml"/>
<include href="inputs/MPM/MaterialData/VonMisesYield.xml"/>
<include href="inputs/MPM/MaterialData/DruckerBeckerStabilityCheck.xml"/>
<include href="inputs/MPM/MaterialData/JCDamage4340St.xml"/>
<specific_heat_model type="steel_Cp"> </specific_heat_model>

<initial_mean_porosity> 0.005 </initial_mean_porosity>
<initial_std_porosity> 0.001 </initial_std_porosity>
<critical_porosity> 0.3 </critical_porosity>
<frac_nucleation> 0.1 </frac_nucleation>
<meanstrain_nucleation> 0.3 </meanstrain_nucleation>
<stddevstrain_nucleation> 0.1 </stddevstrain_nucleation>
<initial_porosity_distrib> gauss </initial_porosity_distrib>

<initial_mean_scalar_damage> 0.005 </initial_mean_scalar_damage>
<initial_std_scalar_damage> 0.001 </initial_std_scalar_damage>
<critical_scalar_damage> 1.0 </critical_scalar_damage>
<initial_scalar_damage_distrib> gauss </initial_scalar_damage_distrib>
</constitutive_model>

  <geom_object>
    <difference>
      <cylinder label = "outer cylinder">
        <bottom> [0.0,0.0,-.05715] </bottom>
        <top> [0.0,0.0, .05715] </top>
        <radius> 0.05715 </radius>
      </cylinder>
      <cylinder label = "inner cylinder">
        <bottom> [0.0,0.0,-.0508] </bottom>
        <top> [0.0,0.0, .0508] </top>
        <radius> 0.0508 </radius>
      </cylinder>
    </difference>
    <res> [2,2,2] </res>
    <velocity> [0.0,0.0,0.0] </velocity>
  </geom_object>

```

```

        <temperature>                600                </temperature>
    </geom_object>
</material>
<material name = "reactant">
    <include href="inputs/MPM/MaterialData/MatConstPBX9501.xml"/>
    <constitutive_model type = "visco_scam">
        <include href="inputs/MPM/MaterialData/ViscoSCRAMPBX9501.xml"/>
        <include href="inputs/MPM/MaterialData/TimeTempPBX9501.xml"/>
        <randomize_parameters>                false </randomize_parameters>
        <use_time_temperature_equation> true </use_time_temperature_equation>
        <useObjectiveRate>                    true </useObjectiveRate>
        <useModifiedEOS>                    true </useModifiedEOS>
    </constitutive_model>
    <geom_object>
        <difference>
            <cylinder label = "inner cylinder"> </cylinder>
            <cylinder label = "inner hole">
                <bottom>                [0.0,0.0,-.0508] </bottom>
                <top>                    [0.0,0.0, .0508] </top>
                <radius>                0.01 </radius>
            </cylinder>
        </difference>
        <res>                            [2,2,2] </res>
        <velocity>                       [0.0,0.0,0.0] </velocity>
        <temperature>                   440.0 </temperature>
    </geom_object>
</material>

    <contact>
        <type>approach</type>
        <materials>                      [0,1] </materials>
        <mu> 0.0 </mu>
    </contact>
    <thermal_contact>
    </thermal_contact>
</MPM>

<ICE>
    <material>
        <EOS type = "ideal_gas">
        </EOS>
        <dynamic_viscosity>              0.0 </dynamic_viscosity>
        <thermal_conductivity>           0.0 </thermal_conductivity>
        <specific_heat>                  716.0 </specific_heat>
        <gamma>                          1.4 </gamma>
        <geom_object>
            <difference>

```

```

        <box>
            <min>                [-0.254,-0.254,-0.254] </min>
            <max>                [ 0.254, 0.254, 0.254] </max>
        </box>
        <cylinder label = "outer cylinder"> </cylinder>
    </difference>
    <cylinder label="inner hole"> </cylinder>
    <res>                        [2,2,2]                </res>
    <velocity>                   [0.0,0.0,0.0]           </velocity>
    <!--
    <temperature>                300.0                   </temperature>
    <density>                    1.1792946927374306000e+00 </density>
    -->
    <temperature>                400.0                   </temperature>
    <density>                    0.884471019553073        </density>
    <pressure>                   101325.0                 </pressure>
    </geom_object>
</material>
</ICE>

<exchange_properties>
    <exchange_coefficients>
        <momentum> [0, 1e15, 1e15]    </momentum>
        <heat>     [0, 1e10, 1e10]    </heat>
    </exchange_coefficients>
</exchange_properties>
</MaterialProperties>

<AddMaterialProperties>
    <ICE>
        <material name = "product">
            <EOS type = "ideal_gas">
            </EOS>
            <dynamic_viscosity>        0.0                </dynamic_viscosity>
            <thermal_conductivity>     0.0                </thermal_conductivity>
            <specific_heat>            716.0              </specific_heat>
            <gamma>                    1.4                </gamma>
            <geom_object>
                <box>
                    <min>                [ 1.0, 1.0, 1.0] </min>
                    <max>                [ 2.0, 2.0, 2.0] </max>
                </box>
                <res>                    [2,2,2]          </res>
                <velocity>               [0.0,0.0,0.0]    </velocity>
                <temperature>            300.0            </temperature>
                <density>                1.1792946927374306000e+00 </density>
                <pressure>               101325.0          </pressure>
            </geom_object>
        </material>
    </ICE>
</AddMaterialProperties>

```

```

        </geom_object>
    </material>
</ICE>

<exchange_properties>
    <exchange_coefficients>
        <momentum> [0, 1e15, 1e15, 1e15, 1e15, 1e15]    </momentum>
        <heat>      [0, 1e10, 1e10, 1e10, 1e10, 1e10]    </heat>
        <!--
        <heat>      [0, 1, 1, 1, 1, 1]    </heat>
        -->
    </exchange_coefficients>
</exchange_properties>
</AddMaterialProperties>

<Models>
    <Model type="Simple_Burn">
        <Active>      false    </Active>
        <fromMaterial> reactant  </fromMaterial>
        <toMaterial>   product   </toMaterial>
        <ThresholdTemp> 450.0    </ThresholdTemp>
        <ThresholdPressure> 50000.0 </ThresholdPressure>
        <Enthalpy>      2000000.0 </Enthalpy>
        <BurnCoeff>      75.3    </BurnCoeff>
        <refPressure>    101325.0 </refPressure>
    </Model>
</Models>

</Uintah_specification>

```

The PBS script used to run this test is

```

#
# ASK PBS TO SEND YOU AN EMAIL ON CERTAIN EVENTS: (a)bort (b)egin (e)nd (n)ever
#
# (User May Change)

#PBS -m abe

#
# SET THE NAME OF THE JOB:
#
# (User May Change)

```

```

#PBS -N ExplodeRing

#
# SET THE QUEUE IN WHICH TO RUN THE JOB.
#      (Note, there is currently only one queue, so you should never change this field)

#PBS -q defaultq

#
# SET THE RESOURCES (# NODES, TIME) REQUESTED FROM THE BATCH SCHEDULER:
#   - select: <# nodes>,ncpus=2,walltime=<time>
#   - walltime: walltime before PBS kills our job.
#               [[hours:]minutes:]seconds[.milliseconds]
#               Examples:
#                   walltime=60          (60 seconds)
#                   walltime=10:00      (10 minutes)
#                   walltime=5:00:00    (5 hours)
#
# (User May Change)

#PBS -l select=2:ncpus=2,walltime=24:00

#
# START UP LAM

cd $PBS_O_WORKDIR
lamboot

# [place your command here] >& ${PBS_O_WORKDIR}/output.${PBS_JOBID}
mpirun -np 4 ../sus_opt explodeRFull.ups >& output.${PBS_JOBID}

#
# REMEMBER, IF YOU ARE RUNNING TWO SERIAL JOBS, YOU NEED A:
# wait

#
# STOP LAM

lamhalt -v
exit

```

References

- [1] S. Nemat-Nasser. Rate-independent finite-deformation elastoplasticity: a new explicit constitutive algorithm. *Mech. Mater.*, 11:235–249, 1991.

- [2] S. Nemat-Nasser and D. T. Chung. An explicit constitutive algorithm for large-strain, large-strain-rate elastic-viscoplasticity. *Comput. Meth. Appl. Mech. Engrg*, 95(2):205–219, 1992.
- [3] L. H. Wang and S. N. Atluri. An analysis of an explicit algorithm and the radial return algorithm, and a proposed modification, in finite elasticity. *Computational Mechanics*, 13:380–389, 1994.
- [4] P. J. Maudlin and S. K. Schiferl. Computational anisotropic plasticity for high-rate forming applications. *Comput. Methods Appl. Mech. Engrg.*, 131:1–30, 1996.
- [5] M. A. Zocher, P. J. Maudlin, S. R. Chen, and E. C. Flower-Maudlin. An evaluation of several hardening models using Taylor cylinder impact data. In *Proc. , European Congress on Computational Methods in Applied Sciences and Engineering*, Barcelona, Spain, 2000. ECCOMAS.
- [6] J. C. Simo and T. J. R. Hughes. *Computational Inelasticity*. Springer-Verlag, New York, 1998.
- [7] M. L. Wilkins. *Computer Simulation of Dynamic Phenomena*. Springer-Verlag, Berlin, 1999.
- [8] D. J. Steinberg, S. G. Cochran, and M. W. Guinan. A constitutive model for metals applicable at high-strain rate. *J. Appl. Phys.*, 51(3):1498–1504, 1980.
- [9] L. Burakovsky, D. L. Preston, and R. R. Silbar. Analysis of dislocation mechanism for melting of elements: pressure dependence. *J. Appl. Phys.*, 88(11):6294–6301, 2000.
- [10] Y. P. Varshni. Temperature dependence of the elastic constants. *Physical Rev. B*, 2(10):3952–3958, 1970.
- [11] S. R. Chen and G. T. Gray. Constitutive behavior of tantalum and tantalum-tungsten alloys. *Metall. Mater. Trans. A*, 27A:2994–3006, 1996.
- [12] M.-H. Nadal and P. Le Poac. Continuous model for the shear modulus as a function of pressure and temperature up to the melting point: analysis and ultrasonic validation. *J. Appl. Phys.*, 93(5):2472–2480, 2003.
- [13] G. R. Johnson and W. H. Cook. A constitutive model and data for metals subjected to large strains, high strain rates and high temperatures. In *Proc. 7th International Symposium on Ballistics*, pages 541–547, 1983.
- [14] D. J. Steinberg and C. M. Lund. A constitutive model for strain rates from 10^{-4} to 10^6 s^{-1} . *J. Appl. Phys.*, 65(4):1528–1533, 1989.
- [15] K. G. Hoge and A. K. Mukherjee. The temperature and strain rate dependence of the flow stress of tantalum. *J. Mater. Sci.*, 12:1666–1672, 1977.
- [16] F. J. Zerilli and R. W. Armstrong. Dislocation-mechanics-based constitutive relations for material dynamics calculations. *J. Appl. Phys.*, 61(5):1816–1825, 1987.
- [17] F. J. Zerilli and R. W. Armstrong. Constitutive relations for the plastic deformation of metals. In *High-Pressure Science and Technology - 1993*, pages 989–992, Colorado Springs, Colorado, 1993. American Institute of Physics.
- [18] F. J. Zerilli. Dislocation mechanics-based constitutive equations. *Metall. Mater. Trans. A*, 35A:2547–2555, 2004.

- [19] F. H. Abed and G. Z. Voyiadjis. A consistent modified Zerilli-Armstrong flow stress model for bcc and fcc metals for elevated temperatures. *Acta Mechanica*, 175:1–18, 2005.
- [20] P. S. Follansbee and U. F. Kocks. A constitutive description of the deformation of copper based on the use of the mechanical threshold stress as an internal state variable. *Acta Metall.*, 36:82–93, 1988.
- [21] D. M. Goto, J. F. Bingert, W. R. Reed, and R. K. Garrett. Anisotropy-corrected MTS constitutive strength modeling in HY-100 steel. *Scripta Mater.*, 42:1125–1131, 2000.
- [22] U. F. Kocks. Realistic constitutive relations for metal plasticity. *Materials Science and Engrg.*, A317:181–187, 2001.
- [23] D. L. Preston, D. L. Tonks, and D. C. Wallace. Model of plastic deformation for extreme loading conditions. *J. Appl. Phys.*, 93(1):211–220, 2003.
- [24] G. Ravichandran, A. J. Rosakis, J. Hodowany, and P. Rosakis. On the conversion of plastic work into heat during high-strain-rate deformation. In *Proc. , 12th APS Topical Conference on Shock Compression of Condensed Matter*, pages 557–562. American Physical Society, 2001.
- [25] F. L. Lederman, M. B. Salamon, and L. W. Shacklette. Experimental verification of scaling and test of the universality hypothesis from specific heat data. *Phys. Rev. B*, 9(7):2981–2988, 1974.
- [26] J. O. Coplien. *Advanced C++ Programming Styles and Idioms*. Addison-Wesley, Reading, MA, 1992.
- [27] B. Banerjee. Material point method simulations of fragmenting cylinders. In *Proc. 17th ASCE Engineering Mechanics Conference (EM2004)*, Newark, Delaware, 2004.
- [28] B. Banerjee. MPM validation: Sphere-cylinder impact: Low resolution simulations. Technical Report C-SAFE-CD-IR-04-002, Center for the Simulation of Accidental Fires and Explosions, University of Utah, USA, 2004.
- [29] B. Banerjee. Simulation of impact and fragmentation with the material point method. In *Proc. 11th International Conference on Fracture*, Turin, Italy, 2005.
- [30] A. L. Gurson. Continuum theory of ductile rupture by void nucleation and growth: Part 1. Yield criteria and flow rules for porous ductile media. *ASME J. Engg. Mater. Tech.*, 99:2–15, 1977.
- [31] V. Tvergaard and A. Needleman. Analysis of the cup-cone fracture in a round tensile bar. *Acta Metall.*, 32(1):157–169, 1984.
- [32] S. Ramaswamy and N. Aravas. Finite element implementation of gradient plasticity models Part II: Gradient-dependent evolution equations. *Comput. Methods Appl. Mech. Engrg.*, 163:33–53, 1998.
- [33] C. C. Chu and A. Needleman. Void nucleation effects in biaxially stretched sheets. *ASME J. Engg. Mater. Tech.*, 102:249–256, 1980.
- [34] G. R. Johnson and W. H. Cook. Fracture characteristics of three metals subjected to various strains, strain rates, temperatures and pressures. *Int. J. Eng. Fract. Mech.*, 21:31–48, 1985.
- [35] J. N. Johnson and F. L. Addessio. Tensile plasticity and ductile fracture. *J. Appl. Phys.*, 64(12):6699–6712, 1988.
- [36] D. C. Drucker. A definition of stable inelastic material. *J. Appl. Mech.*, 26:101–106, 1959.

- [37] J. W. Rudnicki and J. R. Rice. Conditions for the localization of deformation in pressure-sensitive dilatant materials. *J. Mech. Phys. Solids*, 23:371–394, 1975.
- [38] P. Perzyna. Constitutive modelling of dissipative solids for localization and fracture. In Perzyna P., editor, *Localization and Fracture Phenomena in Inelastic Solids: CISM Courses and Lectures No. 386*, pages 99–241. SpringerWien, New York, 1998.
- [39] R. Becker. Ring fragmentation predictions using the gurson model with material stability conditions as failure criteria. *Int. J. Solids Struct.*, 39:3555–3580, 2002.
- [40] R. Hill and J. W. Hutchinson. Bifurcation phenomena in the plane tension test. *J. Mech. Phys. Solids*, 23:239–264, 1975.
- [41] Z. P. Bazant and T. Belytschko. Wave propagation in a strain-softening bar: Exact solution. *ASCE J. Engg. Mech*, 111(3):381–389, 1985.
- [42] V. Tvergaard and A. Needleman. Ductile failure modes in dynamically loaded notched bars. In J. W. Ju, D. Krajcinovic, and H. L. Schreyer, editors, *Damage Mechanics in Engineering Materials: AMD 109/MD 24*, pages 117–128. American Society of Mechanical Engineers, New York, NY, 1990.
- [43] S. Ramaswamy and N. Aravas. Finite element implementation of gradient plasticity models Part I: Gradient-dependent yield functions. *Comput. Methods Appl. Mech. Engrg.*, 163:11–32, 1998.
- [44] S. Hao, W. K. Liu, and D. Qian. Localization-induced band and cohesive model. *J. Appl. Mech.*, 67:803–812, 2000.