

# Thermally-assisted generation of Single-Photon Emitters in monolayer WS<sub>2</sub>

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## Abstract

The controlled generation of single-photon emitters (SPEs) in two-dimensional (2D) semiconductors is an essential step towards their integration in quantum communication and information processing platforms. Here, we use *in situ* high-temperature annealing of hBN-encapsulated monolayer WS<sub>2</sub> on a suspended micro-heater to produce spectrally narrow single-photon emitters at cryogenic temperatures. Annealing at 1100 K leads to the emergence of a sharp, nanosecond-lived emission line  $X_L$  whose linewidth is at least as small as 200  $\mu$ eV, and which is red-shifted by 80 meV from the neutral exciton emission. Photoluminescence excitation spectroscopy and power-dependent photoluminescence confirms that  $X_L$  originates from annealing-induced defects in the WS<sub>2</sub> monolayer. Second-order photon correlation measurements show clear antibunching with  $g^{(2)}(0) < 0.5$ , confirming the single character of the observed defect. This approach opens the door for a controlled and robust pathway for engineering SPEs in monolayer WS<sub>2</sub> and adds a new member to the ever-growing family of defect-based quantum light sources in 2D semiconductors.

## Introduction

The reduced dielectric screening and strong quantum confinement in monolayer transition metal dichalcogenides (TMDs) lead to a remarkable enhancement of the Coulomb interaction [1], such that excitons dominate their optical properties even at room temperature [2]. Owing to their two-dimensional (2D) nature and small radius [3], excitons in TMDs and other 2D materials are highly sensitive to local perturbations. As a result, defects, strain, and dielectric disorder can efficiently confine excitons [4], enabling the engineering of point-like defects for the generation of single-photon emitters in atomically-thin layers.

Over the past decade, intense efforts have been devoted to understanding and controlling single-photon emission in 2D semiconductors [5]. Following the first observations of single-photon emitters in monolayer WSe<sub>2</sub> [6, 7], a variety of deterministic approaches have emerged, mainly based on strain engineering and electron/ion irradiation. Focused helium ion irradiation has enabled the creation of spatially ordered arrays of single-photon emitters in monolayer MoS<sub>2</sub> with nanometric precision [8, 9], although the efficiency of this method is significantly reduced in the similar material WS<sub>2</sub> [10]. Electron-beam irradiation has also been employed to activate individual colour centers in hexagonal boron nitride with high lateral precision [11]. Alternatively, thermal annealing under controlled environments has recently been shown to generate defect-related emission in monolayer MoS<sub>2</sub> [12] and ultra-sharp green emitters in hBN under oxygen atmosphere [13].

In this work, we adopt this latter approach and demonstrate the generation of single-photon emitters in hBN-encapsulated WS<sub>2</sub> through high-vacuum thermal annealing. We fabricate hBN/WS<sub>2</sub>/hBN heterostructures on suspended SiC micro-membranes that act as local micro-heaters. Integration into a cryostat enables Joule heating of the membrane up to  $\sim 1200$  K while maintaining the surrounding environment at cryogenic temperature. We show that this controlled thermal treatment leads to the formation of individual quantum emitters in monolayer WS<sub>2</sub>. We measure a recombination time in the nanosecond range at cryogenic temperatures and perform second-order photon correlation measurements yielding  $g^{(2)}(0) \approx 0.4$ , confirming the single-photon nature of the emission. Photoluminescence excitation spectroscopy (PLE) further demonstrates that the emitting defects originate from the TMD layer rather than from the hBN or the SiC membrane. The observed single-photon emission is attributed to excitons localized at annealing-induced defect sites, whose microscopic origin remains to be elucidated.

## Results

### Sample preparation and optical characterization

Fully encapsulated WS<sub>2</sub> monolayers were prepared by mechanical exfoliation of bulk crystals followed by the assembly of hBN/WS<sub>2</sub>/hBN heterostructures on Si/SiO<sub>2</sub> substrates through a dry stamp transfer technique [14]. The stack was then picked up using a polycarbonate (PC) stamp and transferred onto 120 nm-thick suspended SiC membranes by thermal release at  $\sim 200^\circ\text{C}$ , the resulting heterostructure is schematically shown in Fig. 1(a). Figure 1(b) shows an optical image under white light illumination of an hBN encapsulated monolayer WS<sub>2</sub> transferred onto a SiC membrane. The latter is suspended between two silicon contacts allowing for a current to be applied across it.

Unless otherwise stated, the laser source was a pulsed supercontinuum laser coupled to a tunable filter (typical FWHM of 2 nm). The laser pulses have a temporal width of 40 ps, repetition rate of 78 MHz, and were focused onto the sample by an apochromatic objective (NA=0.82) producing a spot of  $\sim 1 \mu\text{m}$  diameter. The photoluminescence was collected by the same objective, dispersed by a grating spectrometer and imaged by a Peltier-cooled CCD camera. For time-resolved and photon-correlation measurements, the photoluminescence was filtered by a double spectrometer and focused into a fibered Hanbury-Brown-Twiss setup. The detectors at the outputs were two identical single-photon avalanche photodiodes of temporal resolution close to 30 ps.

Figure 1(c) shows the low temperature photoluminescence (PL) spectrum of one of the devices, showing sharp excitonic features with well-resolved fine structure, including the neutral exciton  $X^0$ , the negatively charged trions  $T^T$  and  $T^S$ , the dark-exciton  $X^D$ , the negatively-charged biexciton  $XX^-$ , as well as several phonon assisted recombination of dark trions. The linewidths, of only a few meV, confirm the excellent optical quality of the samples and the effectiveness of encapsulation in suppressing dielectric disorder [15]. These peaks are consistent with those previously reported in the literature [16, 17].

Photoluminescence excitation (PLE) spectroscopy, in which the detection energy is fixed while the excitation energy is scanned at constant power, probes the absorption and relaxation pathways that populate a given emission line. A significant increase in PL intensity indicates that the excitation energy is resonant with an excitonic transition, allowing PLE to map the absorption spectrum linked to that state. Our PLE data in Fig. 1(d), corresponding to the emission of the low-energy part of the spectrum, reveals clear resonances at the A and B excitons and at the 2s excited states of the A exciton and the trion. These distinct optical resonances, intrinsic to pristine WS<sub>2</sub>, will be important for the characterisation of single-photon emitters.

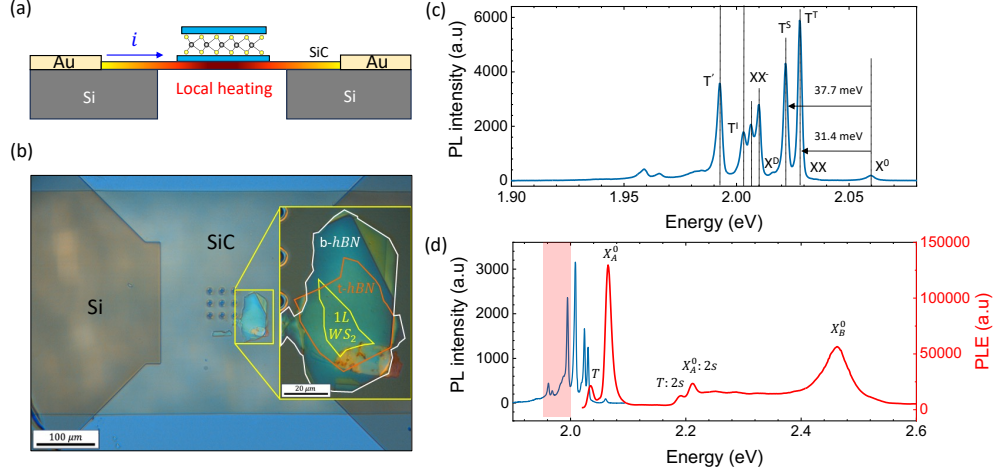


Figure 1: **Optical characterization of hBN-encapsulated monolayer  $\text{WS}_2$  on a micro-heater membrane.** (a) Schematic of the micro-heater chip showing the suspended SiC membrane between metallic contacts and the hBN/ $\text{WS}_2$ /hBN stack transferred on top. (b) Optical image of a representative encapsulated flake on the membrane. (c) Low-temperature PL spectrum at 3.6K displaying sharp neutral-exciton, trion, and dark-state related resonances with linewidths of a few meV ( $\lambda_{\text{laser}} = 514.5$  nm (CW), power  $P_{\text{laser}} = 1$   $\mu\text{W}$ ). (d) PLE spectrum of the highlighted low-energy region exhibiting well-resolved A- and B-exciton absorption resonances and excited excitonic states ( $P_{\text{laser}} = 4$   $\mu\text{W}$ ).

## Thermal annealing and emergence of single-photon emitters

Thermal annealing was performed *in situ* by driving a dc current through the suspended SiC membrane inside the cryostat. In order to determine the maximum annealing temperature above which all spectral signatures of  $\text{WS}_2$  are destroyed, we performed Raman spectroscopy on an already calibrated micro-membrane. We found that the Raman features of monolayer  $\text{WS}_2$  survive up to  $\sim 1220$  K (see Fig. S1 of the Supporting Information), but after 1273 K we did not detect any measurable Raman signature and the monolayer disappeared under white-light illumination. This is in agreement with the early work [18] of Brainard who reported the dissociation of  $\text{WS}_2$  at approximately 1313 K under high vacuum. Moreover, the  $A_{1g}$  mode is spectrally isolated and so its position can be accurately determined at each temperature. The resulting temperature dependence is shown in Fig. S2 and exhibits a clear linear behaviour up to  $950^\circ\text{C}$ , with a shift of  $-0.0134\text{ cm}^{-1}/\text{K}$ , providing an independent and reliable method for temperature calibration.

For each annealing temperature, the current was gradually increased in steps and held for typically 30 min at the desired value, then turned off gradually after which the membrane cooled back to the base temperature of a few kelvin. Importantly, we have checked that during this temperature cycle the membrane does not move significantly and the target sample is still in focus (Fig. S4). This allows us to probe the optical response of the same spot in the monolayer after several annealing steps. As mentioned earlier, the temperature of the sample can be extracted from the position of the  $A_{1g}$  Raman peak. However, a good signal to noise ratio requires long acquisition times, specially at high temperatures. Instead, for the temperature calibration we have measured the PL spectrum at each annealing step as shown in Fig. 2(a) and used Pässler's model [19]:

$$E_g(T) = E_g(0) - \frac{\alpha\Theta}{2} \left[ \sqrt[p]{1 + \left(\frac{2T}{\Theta}\right)^p} - 1 \right]$$

to extract the temperature from the peak position  $E_g(T)$ , indicated by the arrows in Fig. 2(a). We have first validated, using an already temperature-calibrated chip, that Pässler's formula correctly predicts the peak position up to  $\sim 900$  K with the exact same parameters ( $\alpha \approx 3.47 \cdot 10^{-4}\text{ eV}/\text{K}$ ,  $\Theta \approx 200.89\text{ K}$  and

$p \approx 2.35$ ) previously reported for WS<sub>2</sub> [20], except for the exciton position at 0 K which is different in encapsulated monolayers due to the change of the dielectric environment ( $E_g(0) \approx 2.047\text{eV}$  for the sample shown in Fig. 2). For temperatures higher than 950 K, the blackbody radiation of the membrane becomes more intense than the PL itself in the region in which both overlap. In that case we use Planck's law to determine the sample temperature. We have checked that at a temperature at which both the PL and the blackbody radiation are discernable, both methods yield the same temperature up to  $\pm 5$  K (see the Supporting Information).

After annealing up to  $\sim 1000$  K, the PL spectra at cryogenic temperature remain dominated by the intrinsic excitonic emission, apart from redshifts and intensity changes which could be attributed to strain induced by the annealing treatment. After annealing around 1100 K, however, an ultra sharp emission appears on the low-energy side of  $X^0$ . Figure 2(b,c) show the emergence of such narrow emissions in two different samples. These lines, denoted here by  $X_L$ , were located 75 meV and 86 meV below the neutral exciton, respectively. In contrast to the few meV linewidth of the other excitonic peaks observed in as-exfoliated samples,  $X_L$  exhibits a linewidth below 0.2 meV at low excitation power, limited here by our spectrometer's resolution.

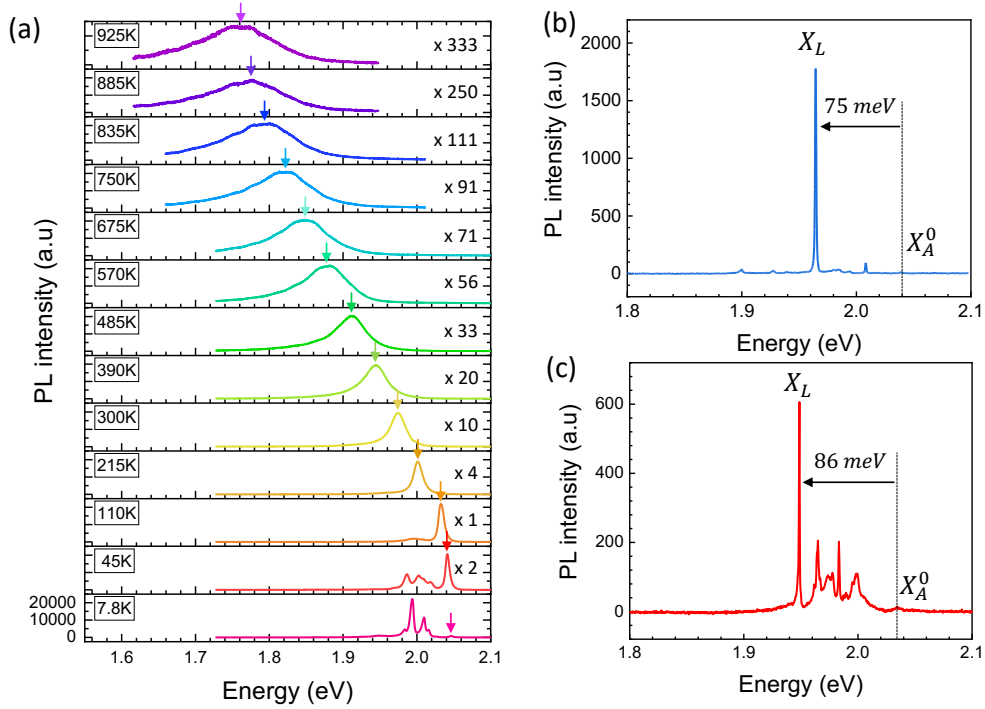


Figure 2: **Thermal-annealing-induced single-photon emission in encapsulated WS<sub>2</sub>.** (a) PL spectra at different temperatures, showing the red-shift of neutral exciton energy with temperature used for temperature calibration. (b, c) PL spectrum after annealing around 1100 K of two samples measured ( $\lambda_{\text{laser}} = 514.5$  nm (CW),  $P_{\text{laser}} = 1$   $\mu\text{W}$ ,  $T = 3.65$  K) in panel (b) and ( $\lambda_{\text{laser}} = 503$  nm (pulsed),  $P_{\text{laser}} = 1$   $\mu\text{W}$ ,  $T = 3.72$  K) in panel (c).

### Excitonic origin and recombination dynamics of $X_L$

We have performed PLE spectroscopy on the  $X_L$  emission while scanning the laser energy across the excitonic resonances. In both samples, the resulting PLE spectrum (shown in Fig. 3(a) for one of them) exhibit the two main absorption resonances intrinsic to monolayer WS<sub>2</sub>: the A and B exciton transitions. This mirrors the PLE recorded in the as-exfoliated WS<sub>2</sub> (Fig. 1(d)), and demonstrates that  $X_L$  emission is enhanced whenever free excitons are generated in the WS<sub>2</sub> layer, excluding other possible origins such as

thermally-induced defects in the hBN encapsulation layers or in the SiC membrane.

In addition, power-dependent PL measurements (shown in Fig. 3(b)) confirm the defect nature of  $X_L$ . Indeed, the integrated intensity increases linearly with excitation power before saturating, as expected for a small number of defect sites that can host at most one exciton at a time. The continuous line in Fig. 3(b) corresponds to a fit of the form:

$$I_{\text{PL}}(P_{\text{avg}}) = C \left( 1 - e^{-\frac{\alpha}{f_{\text{rep}}} P_{\text{avg}}} \right)$$

where  $f_{\text{rep}}$  denotes the laser repetition rate,  $P_{\text{avg}}$  represents the average laser power,  $C$  is a constant, and  $\alpha$  is the fitting parameter. The justification of this model is found in the Supporting Information. The value of  $\alpha$  allows us to estimate that the probability to excite the defect is  $p = 2.13 \times 10^{-6}$  per incident photon. Moreover, the number of counts at saturation allows us to estimate the quantum yield of the emitter. This is because the measured lifetime (to be discussed below) is much longer than the pulse duration, yet much shorter than the interval between pulses. At saturation, at most one photon per pulse can be therefore emitted. By taking into account the losses of the optical setup, we estimate the quantum yield to be  $\sim 5\%$ . Figure 3(b) also shows that the linewidth of  $X_L$  broadens with power, possibly due to local heating caused by the laser.

Figure 3(c) shows the time-resolved PL (TRPL) measurement of  $X_L$  emission. It decays with a characteristic lifetime of  $\sim 0.9$  ns. This lifetime is significantly longer than that of trion emission (typically 100 ps) in the same sample and similar to the lifetime typically reported for single-photon emitters in 2D materials [5, 7]. Finally, the quantum nature of  $X_L$  is probed through second-order photon correlation measurements using a Hanbury Brown-Twiss setup under pulsed excitation. The measured intensity correlation function  $g^{(2)}(\tau)$ , shown in Fig. 3(d), shows a clear antibunching dip at zero delay with  $g^{(2)}(0) \approx 0.4$  without any background correction procedure. This unambiguously proves that  $X_L$  acts as a single-photon emitter. The moderate residual value of  $g^{(2)}(0)$  is attributed to background emission and imperfect spectral filtering, and could be further reduced by improving the collection optics (to decrease the laser excitation power) and spectral selection.

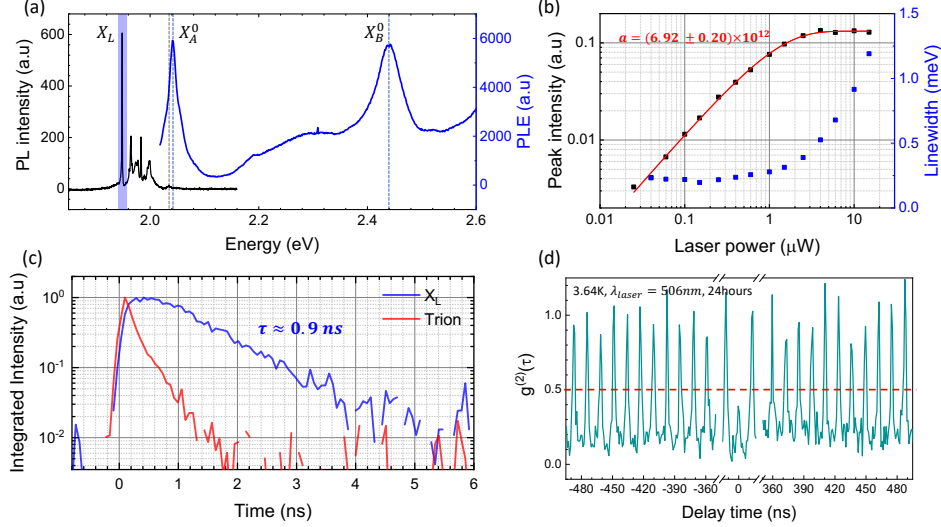


Figure 3: **Optical signatures of the SPE  $X_L$ .** (a) PLE spectrum recorded at the  $X_L$  emission energy, showing strong resonances at the A and B exciton absorption energies. (b) Power dependence of the  $X_L$  intensity and linewidth, exhibiting saturation of the intensity and resolution-limited linewidth at low power. ( $\lambda_{\text{laser}} = 503$  nm (pulsed)). (c) Time-resolved PL traces of  $X_L$  and trion emission, highlighting the longer lifetime of the localized state. ( $\lambda_{\text{laser}} = 506$  nm (pulsed), power  $P_{\text{laser}} = 2.5$   $\mu\text{W}$ ). (d) Second-order correlation function  $g^{(2)}(\tau)$  under pulsed excitation, displaying pronounced antibunching with  $g^{(2)}(0) < 0.5$ . ( $\lambda_{\text{laser}} = 506$  nm (pulsed), power  $P_{\text{laser}} = 1$   $\mu\text{W}$ ).

## Discussion

Although the generation of defects (not necessarily single-photon emitters) by thermal annealing has already been reported in other two-dimensional materials, the high temperature required here (above 1000 K) for the observation of single-photon emitters is uncommon. For example, recent results on monolayer  $\text{MoS}_2$  showed that defect emission attributed to sulfur vacancies already becomes visible after annealing at 500 K, and that the photoluminescence completely vanishes above 900 K [12]. If the nature of the defect observed here is also linked to chalcogen vacancies (or any complex involving such defect), our results can be explained by their significantly higher activation energy in  $\text{WS}_2$  compared to  $\text{MoS}_2$ . This is consistent with the stronger W-S bond compared to the Mo-S bond and with calculations that put the chalcogen vacancy formation energy in  $\text{WS}_2$  as the highest of the TMD family [21]. Our results further show that the photoluminescence of  $\text{WS}_2$  can withstand thermal treatments at substantially higher temperatures than  $\text{MoS}_2$ .

We have also observed another defect-related emission in several of our samples after annealing at 800 K, as shown in the Supporting Information. This line, denoted here by  $X'_L$ , is red-shifted by 120 meV with respect to the neutral exciton and has a 3 meV linewidth at low temperatures (see Fig. S5). This is closer to the redshift typically observed in  $\text{MoS}_2$  and consistent with a recent work that attributes this spectral feature to paired-sulfur vacancies in electron-irradiated  $\text{WS}_2$  [22]. Although we did not observe anti-bunching in photon-correlation experiments for this line, we can't exclude that single-photon emission is possible if only a single of such defect is isolated under the laser spot. The smaller annealing temperature required to observe this line compared to the  $X_L$  may indicate that  $X'_L$  occur due to migration of already present sulfur vacancies, known to be present in large concentrations in as-exfoliated monolayers [23].

It is worth mentioning that, in non-encapsulated monolayers, only this latter type of defect line was observed, this time red-shifted by 220 meV due to a smaller dielectric screening compared to the encapsulated samples. However, the emission was not stable in time nor under pulsed laser excitation. This highlights the importance of encapsulation for the observation and characterisation of these kind of defects. In addition to

limiting chemical contamination (and possibly oxygen passivation of vacancies), hBN encapsulation is also known to suppress photodoping effects [24] and charge noise, which likely contributes to the high temporal stability and narrow linewidth of the observed emitters in encapsulated monolayers. This strongly indicates that encapsulation is a necessary ingredient to stabilize the charge environment of defects and to enable single-photon emission in WS<sub>2</sub>.

Finally, Helium-ion bombardment of encapsulated monolayer WS<sub>2</sub> has been recently shown to produce narrow emission in the near infrared [10],  $\sim 500$  meV below the exciton emission. We did not observe this kind of emission after annealing, and the large energy difference with respect to the neutral exciton may indicate a very different nature for those defects compared to the  $X_L$  and  $X'_L$  discussed here. Note finally a crucial advantage of our *in situ* annealing approach with respect to standard annealing procedures: after each annealing step, any possible adsorbates are efficiently desorbed and cryopumped by the cold surfaces of the cryostat, preventing the thermally-induced defects from interacting with these molecules.

## Conclusion

We have demonstrated a new way of creating single-photon emitters in monolayer WS<sub>2</sub> by *in situ* high-temperature annealing on a micro-heater platform. The clean dielectric environment provided by hBN enables the emergence of an ultra-narrow localized exciton  $X_L$  located  $\sim 80$  meV below the neutral exciton emission after annealing around 1100 K. This emission exhibit a resolution-limited linewidth below 0.2 meV, a recombination lifetime of  $\sim 0.9$  ns, and clear photon antibunching with  $g^{(2)}(0) \approx 0.4$ , confirming its single-photon character. PLE measurements show that  $X_L$  originates from intrinsic excitons captured by thermally generated defects in WS<sub>2</sub>, whose nature merits further investigation. Given *ab initio* studies of intrinsic defects in TMD monolayers, chalcogen-vacancy formation in WS<sub>2</sub> is predicted to require a significantly higher formation energy than in MoS<sub>2</sub>. This is consistent with our observation that single-photon emitters only appear after annealing above 1000 K, contrasting with lower-temperature defect emission in MoS<sub>2</sub>. To unambiguously identify the microscopic structure — vacancy, divacancy-impurity complex or other configuration — one promising route is to perform scanning tunneling luminescence microscopy, allowing atomic-scale imaging and simultaneous optical readout [25]. Furthermore, polarization-resolved photoluminescence (PL) studies of the emitters are required to establish whether they preserve the spin-valley coupling characteristic of TMD excitons, and to assess the possibility of optical spin control. Another interesting direction is to explore the doping dependence of these emitters in gated heterostructures, and their coupling with a spin-valley-polarized electron gas, which can be optically generated in tungsten-based monolayers [26] — potentially allowing electrical and/or optical tuning of the emitter charge state and spin properties.

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## Data Availability

The data that support the findings of this study are available in Zenodo at (to be completed).

## Supporting information

Additional micro-heater calibration procedures and extended datasets can be found in the Supporting Information.

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