



Fritz-
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Exploring the materials space via regularized and symbolic regression (compressed sensing)

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Fritz-Haber-Institut der MPG, Berlin



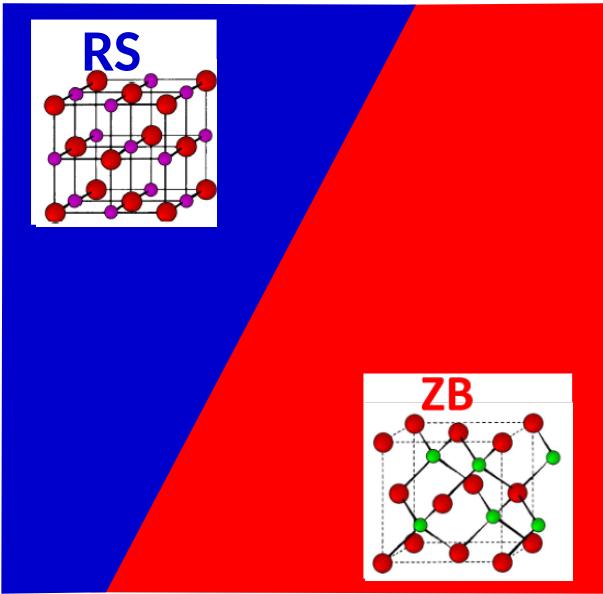
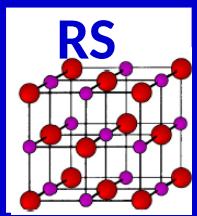
ML4MS 2019
Young Researcher's Workshop on Machine Learning for Materials Science 2019
06-10 May 2019, Aalto University, Helsinki (FI)

Charts/maps of materials



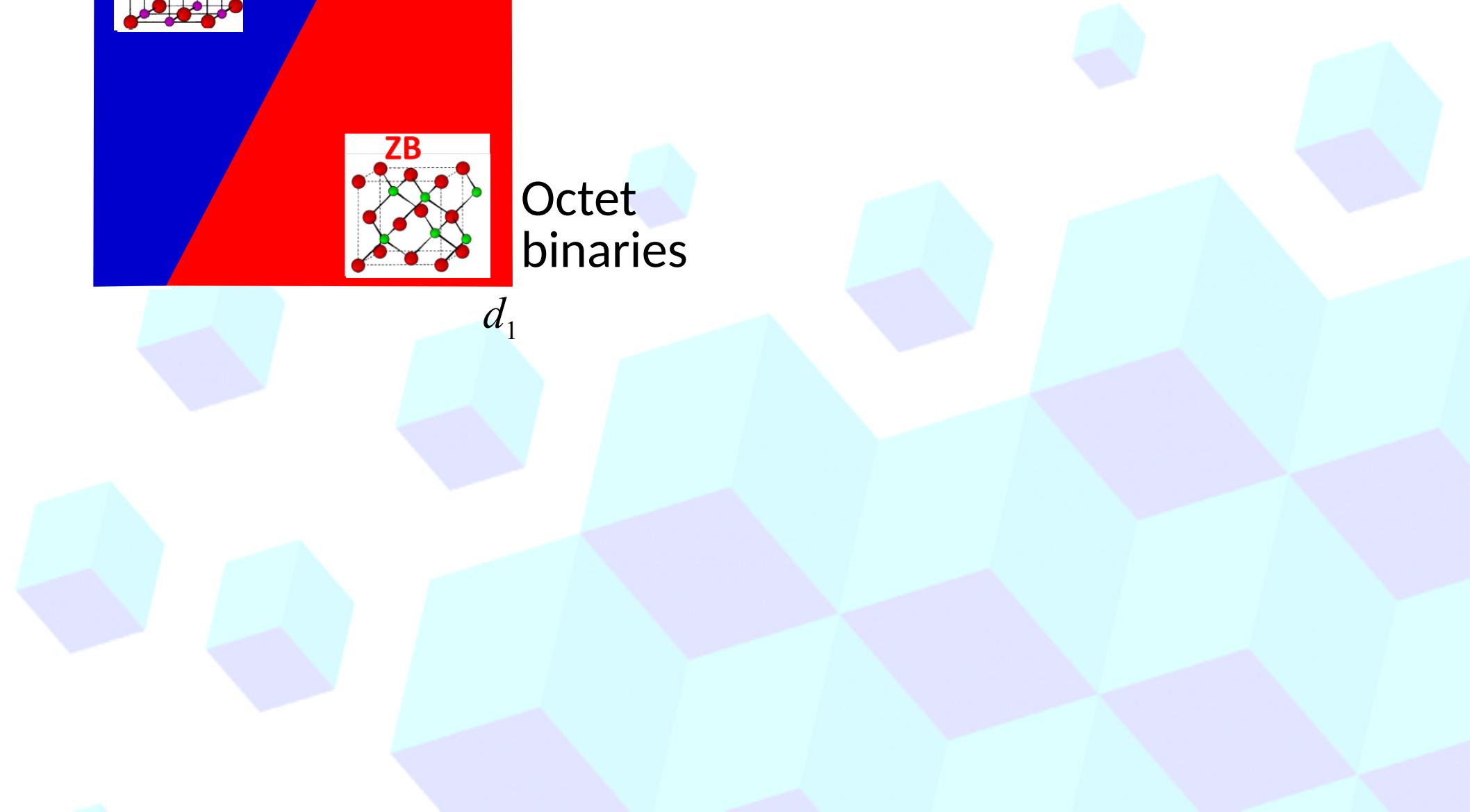
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d_2

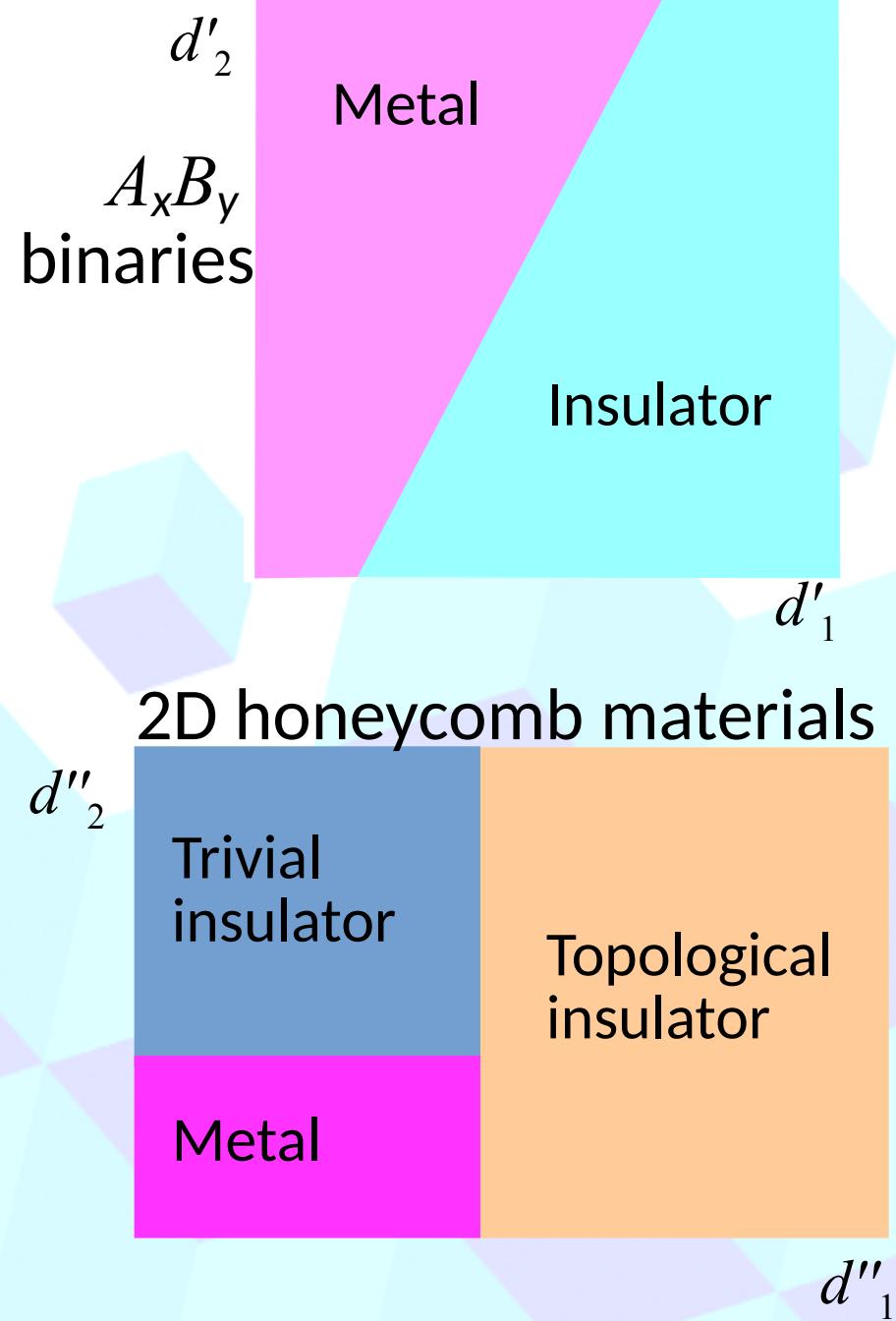
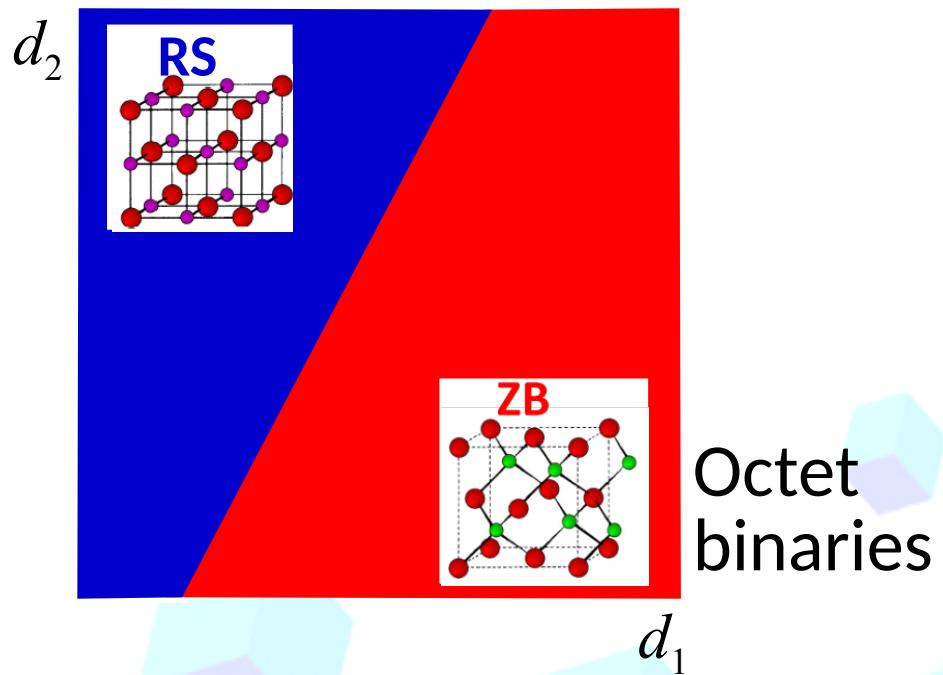


d_1

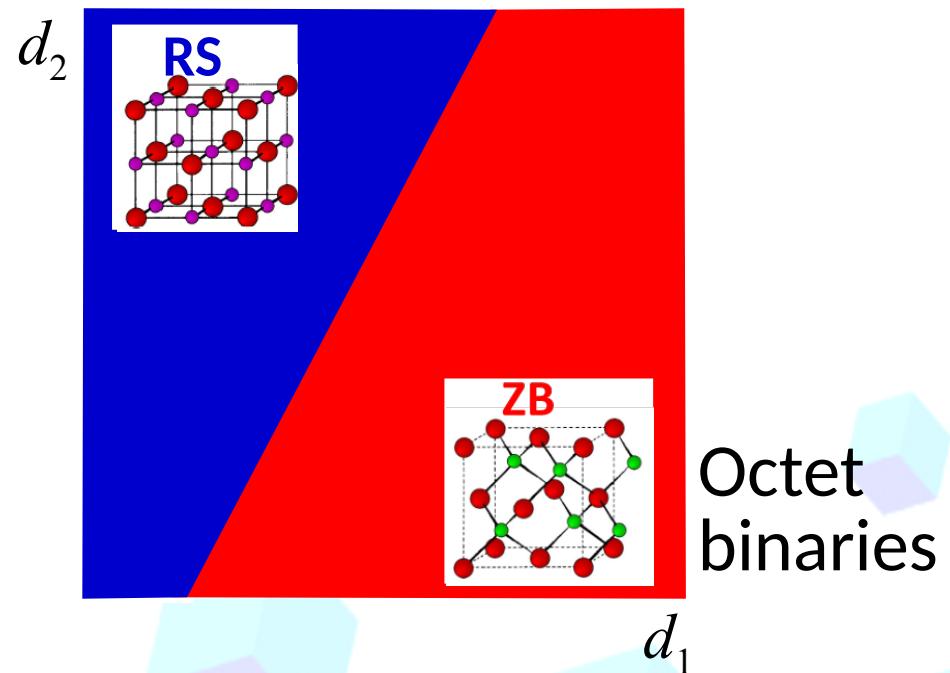
Octet
binaries



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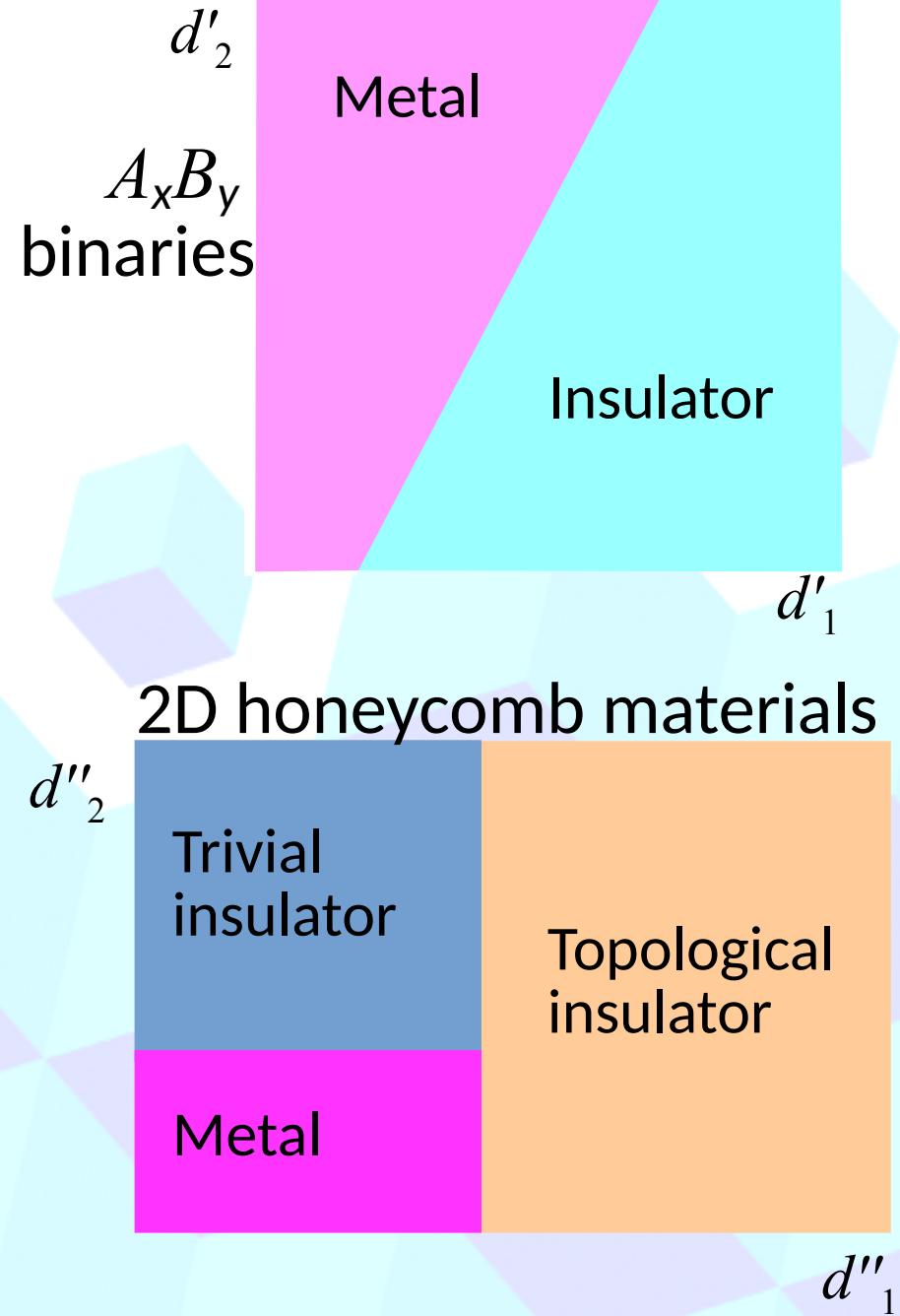
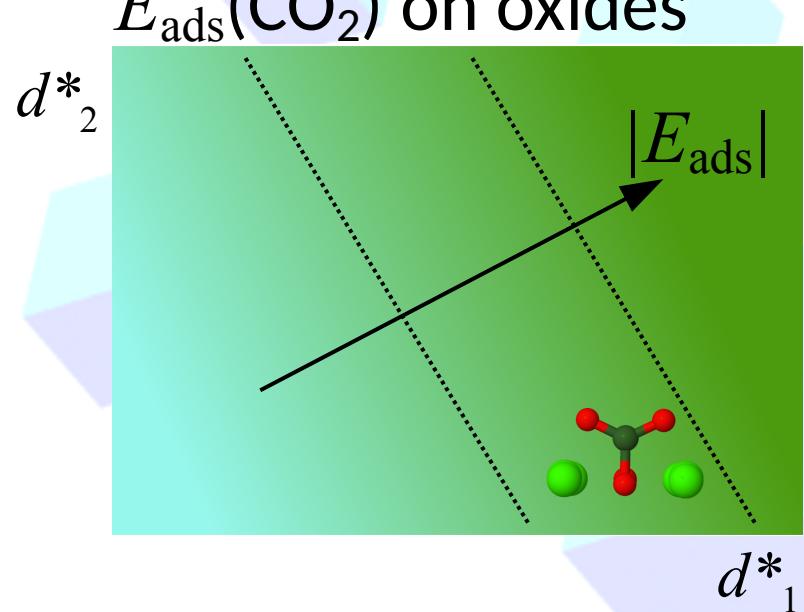


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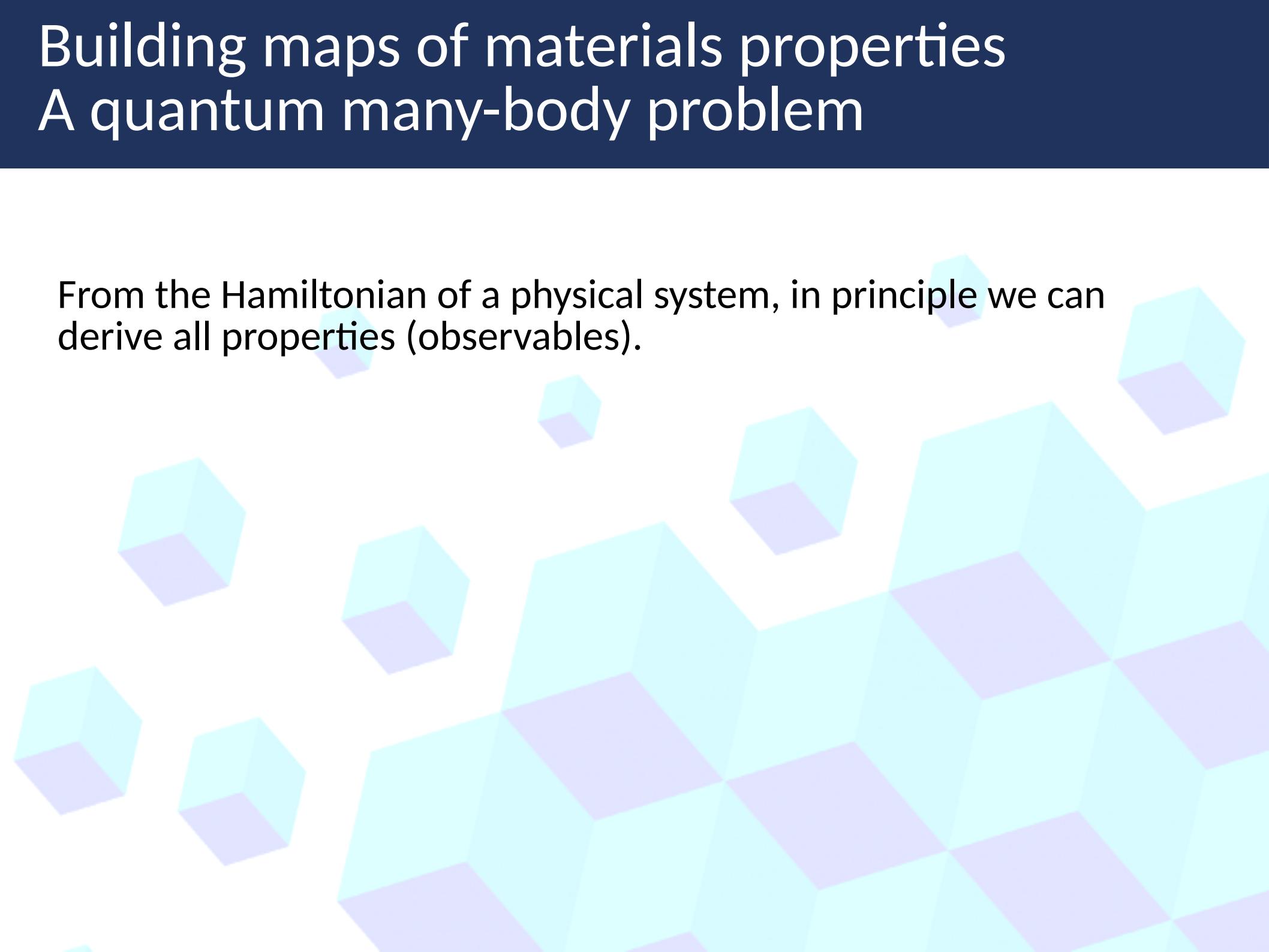
$E_{\text{ads}}(\text{CO}_2)$ on oxides



Building maps of materials properties

A quantum many-body problem

From the Hamiltonian of a physical system, in principle we can derive all properties (observables).



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- what is the most stable crystal structure of each material in the class?

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But in practice, the Hamiltonian is often not the starting point.

For instance, given a class of chemical compositions
(e.g., via prototype formula, such as ABX_3):

- what is the most stable crystal structure of each material in the class?
- which materials are metals / topological insulators / superconductors ?
- which material has the highest melting point?
- which materials has a surface optimal for catalysing some chemical reaction?

The Big Picture

- Design of new materials: preparation, synthesis, and characterization is complex and costly



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The Big Picture

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 - Basic properties determined for very few of them
 - Number of possible materials: practically infinite
- ⇒ New materials with superior properties exist but not yet known
- Data analytics tools will help to identify trends and anomalies in data and guide discovery of new materials

We have a dream

From the **periodic table of the elements**
to **charts of materials**



We have a dream

From the **periodic table of the elements**
to **charts of materials**

Reihen	Gruppe I. — R ² O	Gruppe II. — RO	Gruppe III. — R ² O ³	Gruppe IV. RH ⁴ RO ²	Gruppe V. RH ³ R ² O ⁵	Gruppe VI. RH ² RO ³	Gruppe VII. RH R ² O ⁷	Gruppe VIII. — RO ⁴
1	H=1							
2	Li=7	Be=9.4	B=11	C=12	N=14	O=16	F=19	
3	Na=23	Mg=24	Al=27.3	Si=28	P=31	S=32	Cl=35.5	
4	K=39	Ca=40	—=44	Ti=48	V=51	Cr=52	Mn=55	Fe=56, Co=59, Ni=59, Cu=63.
5	(Cu=63)	Zn=65	—=68	—=72	As=75	Se=78	Br=80	
6	Rb=85	Sr=87	?Yt=88	Zr=90	Nb=94	Mo=96	—=100	Ru=104, Rh=104, Pd=106, Ag=108.
7	(Ag=108)	Cd=112	In=113	Sn=118	Sb=122	Te=125	J=127	
8	Cs=133	Ba=137	?Di=138	?Ce=140	—	—	—	— — — —
9	(—)	—	—	—	—	—	—	—
10	—	—	?Er=178	?La=180	Ta=182	W=184	—	Os=195, Ir=197, Pt=198, Au=199.
11	(Au=199)	Hg=200	Tl=204	Pb=207	Bi=208	—	—	—
12	—	—	—	Th=231	—	U=240	—	—

Mendeleev's 1871 periodic table

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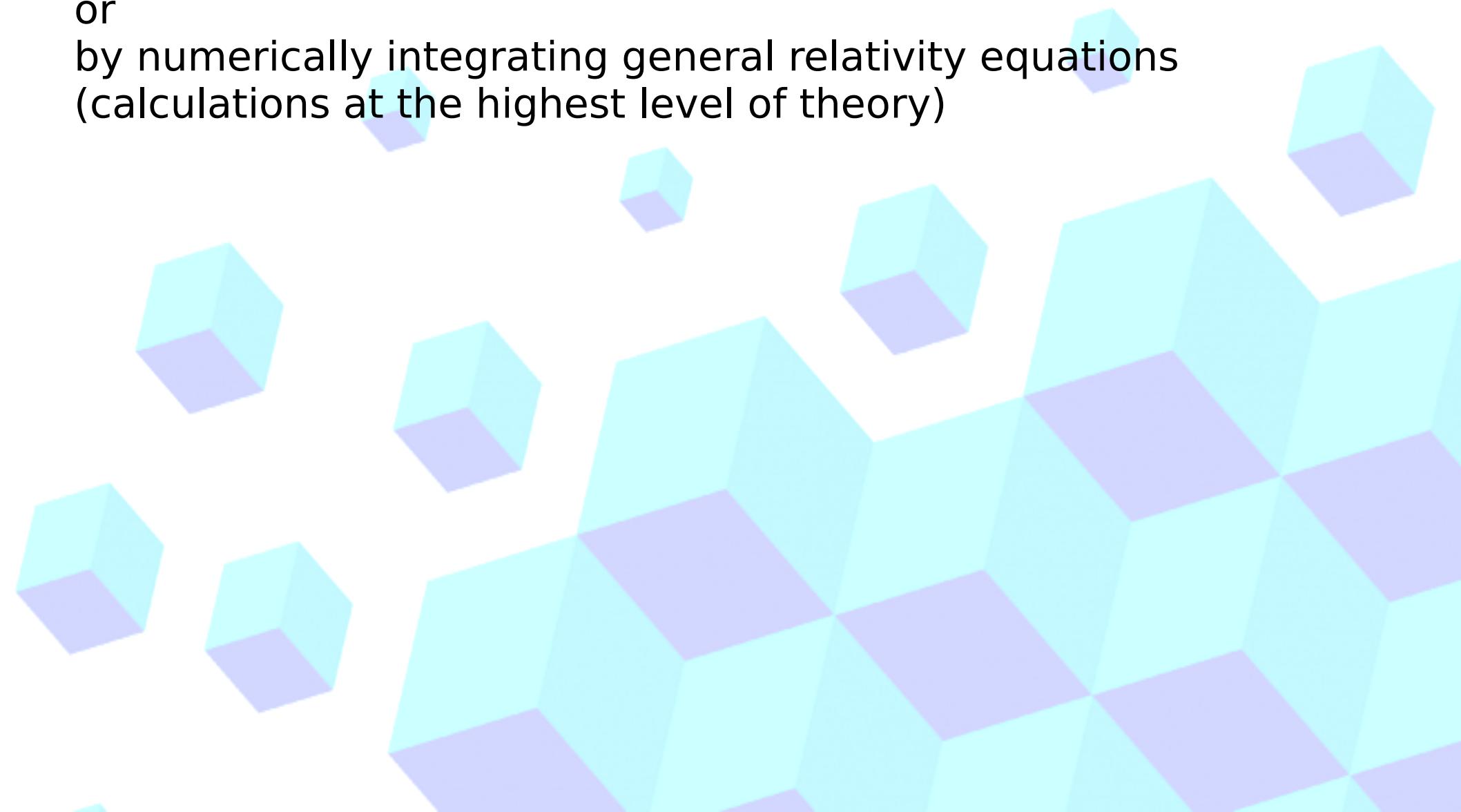
Learning → Discovery

Suppose

to know the trajectories of all planets in the solar system,
from accurate observations (experiment)

or

by numerically integrating general relativity equations
(calculations at the highest level of theory)



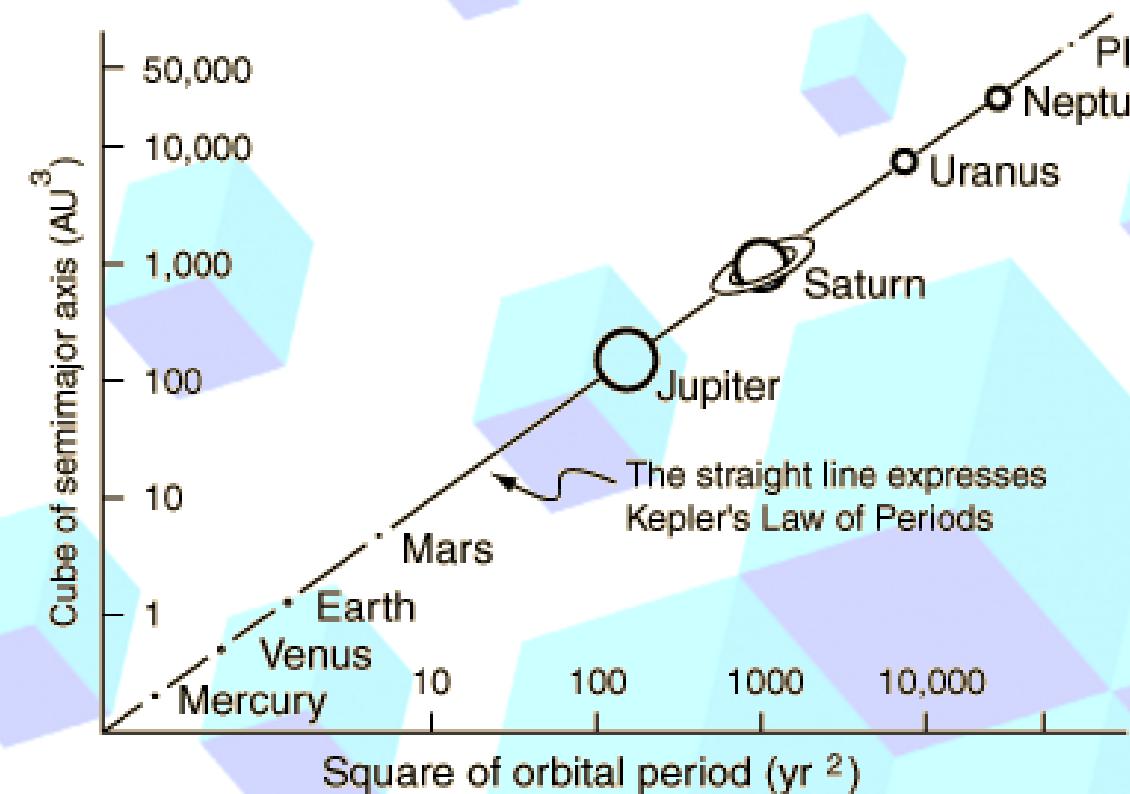
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$$(\text{Orbital period})^2 = C (\text{orbit's major axis})^3$$

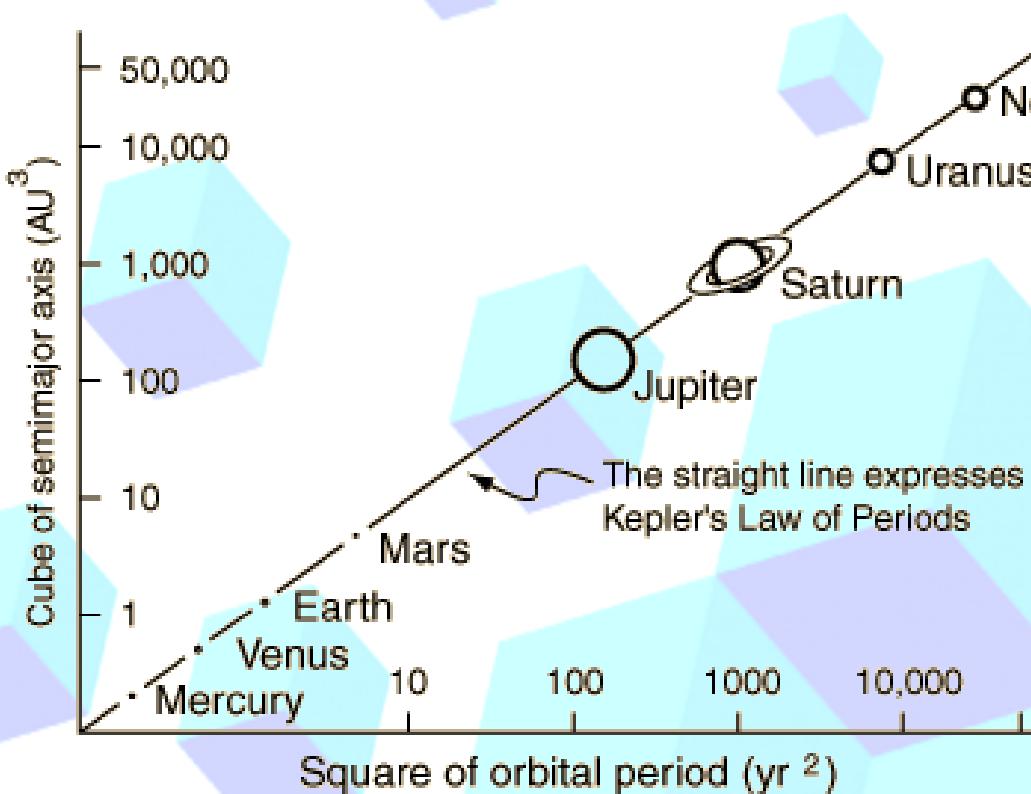
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Data (collected by Tycho Brahe)

Statistical learning (performed by Johannes Kepler)

Physical law (assessed by Isaac Newton)

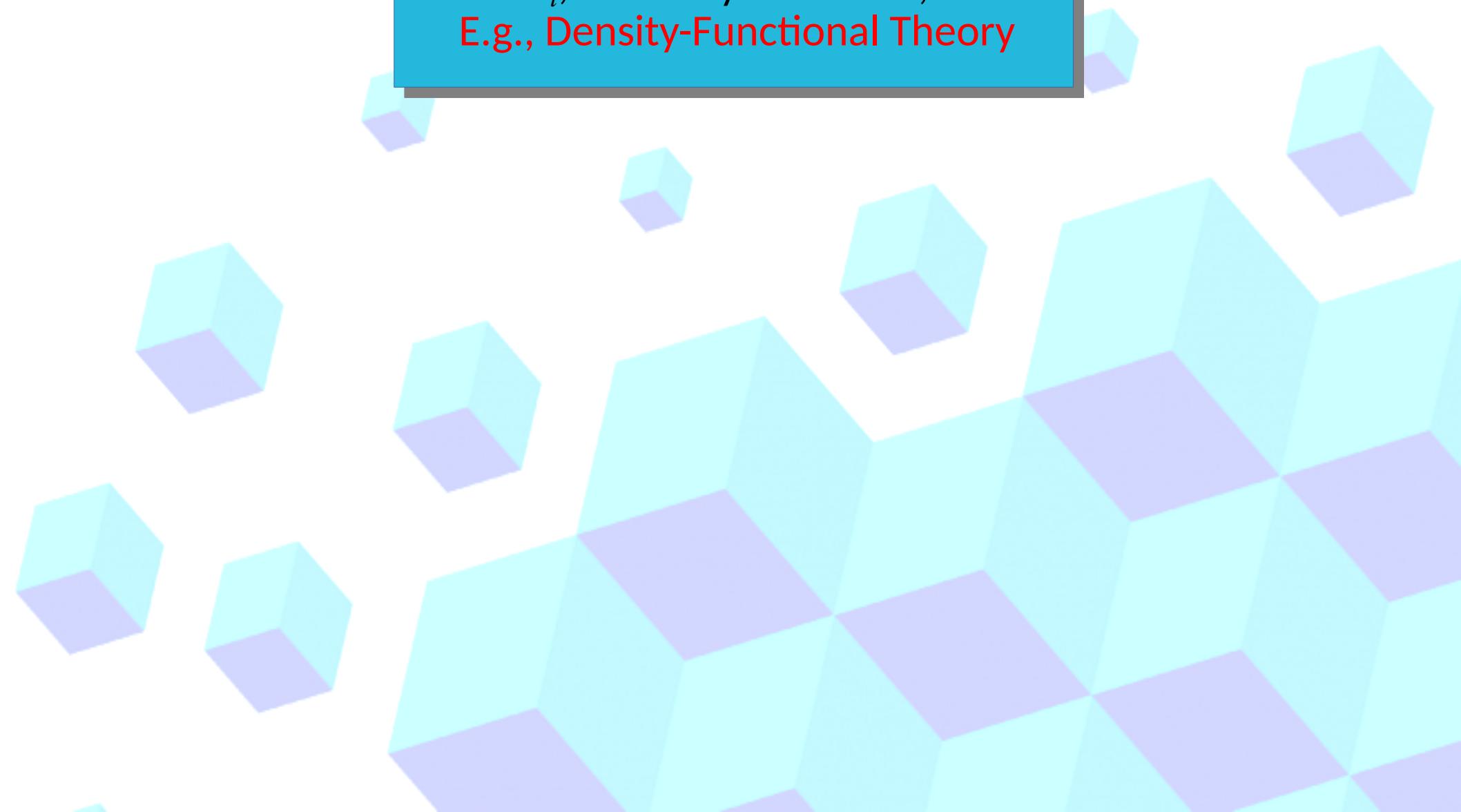
Supervised (big-)data analysis: a flow chart

Training set

Calculate properties and functions

P_i , for many materials, i

E.g., Density-Functional Theory



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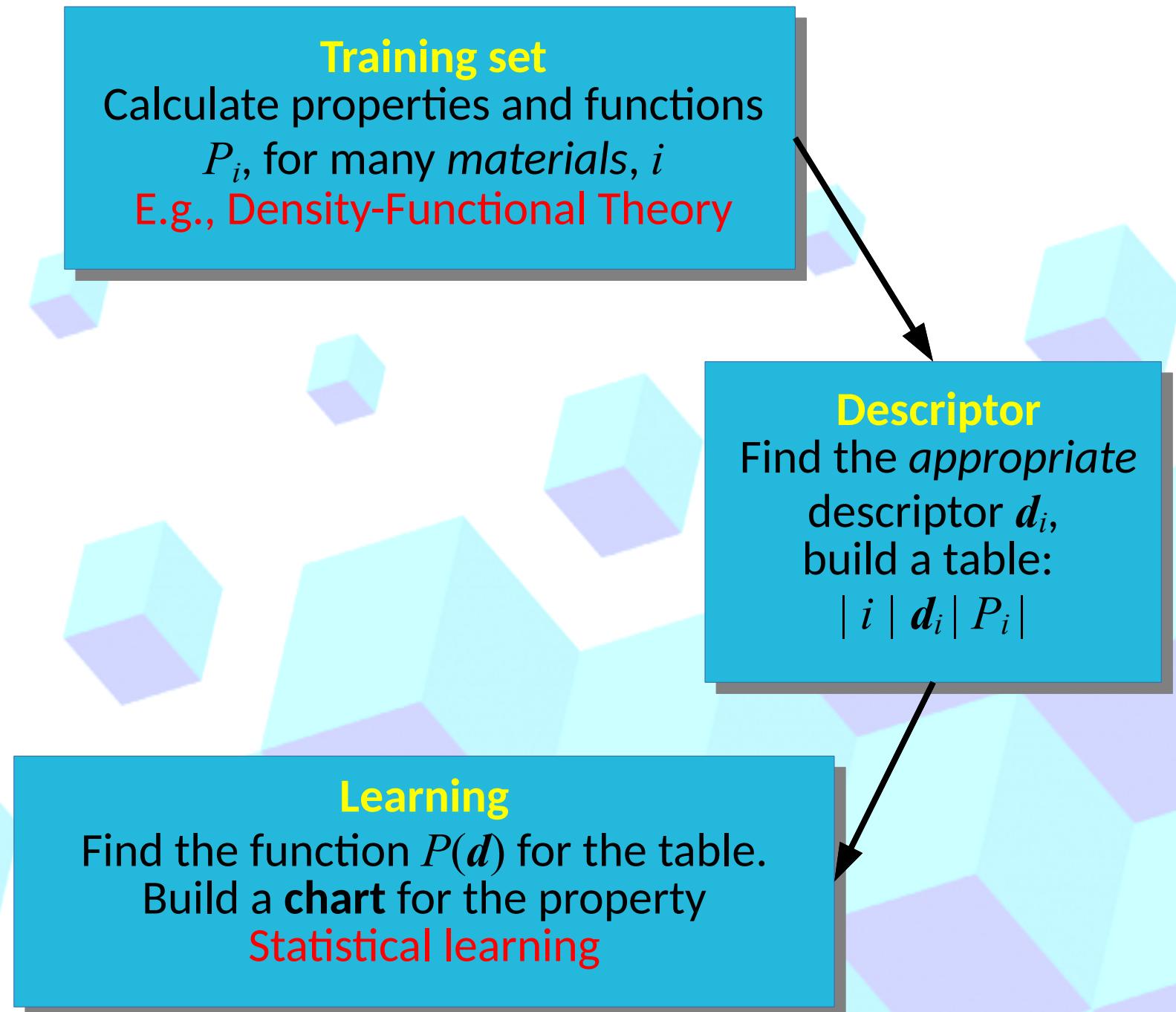
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Descriptor

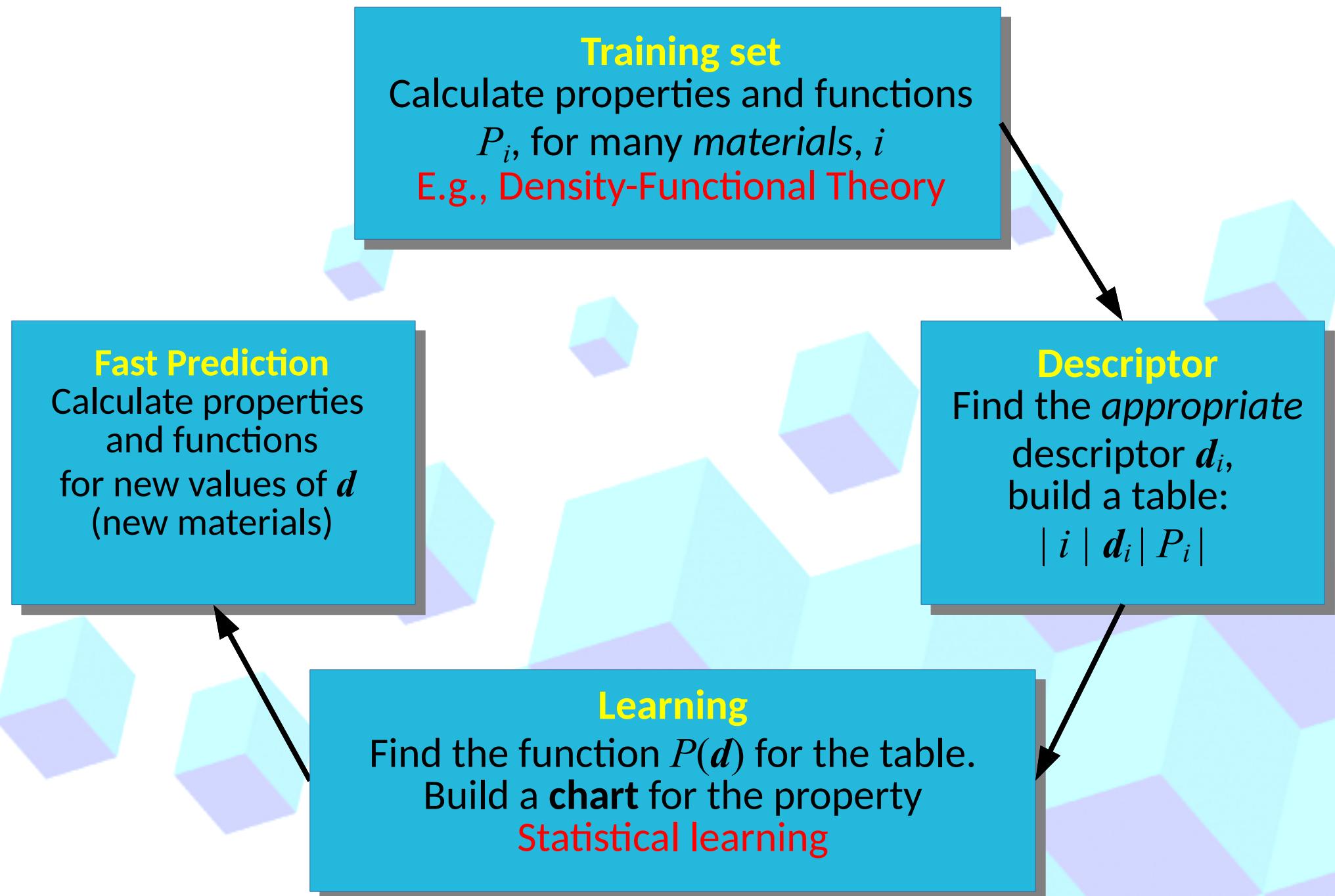
Find the *appropriate* descriptor d_i ,
build a table:

| i | d_i | P_i |

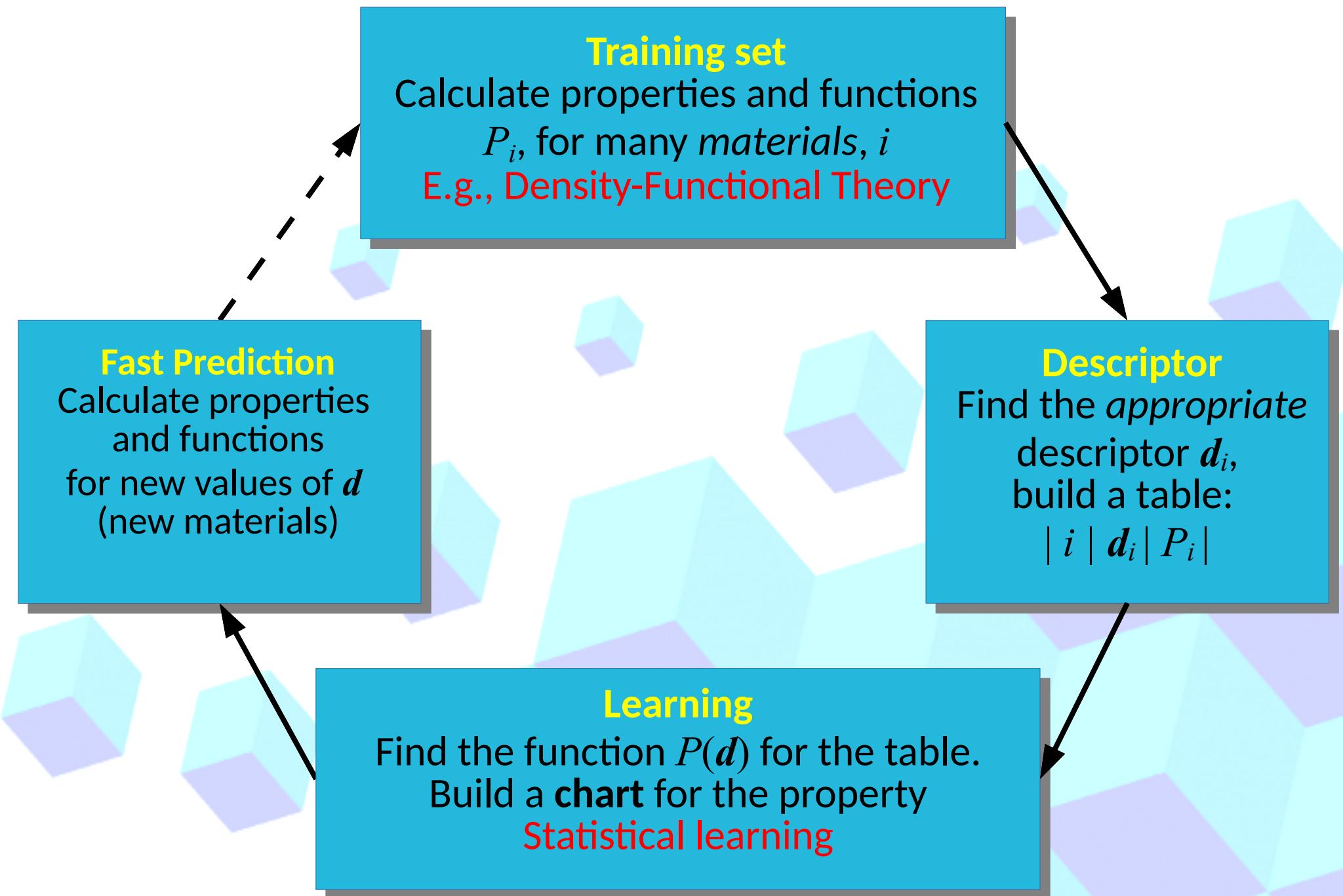
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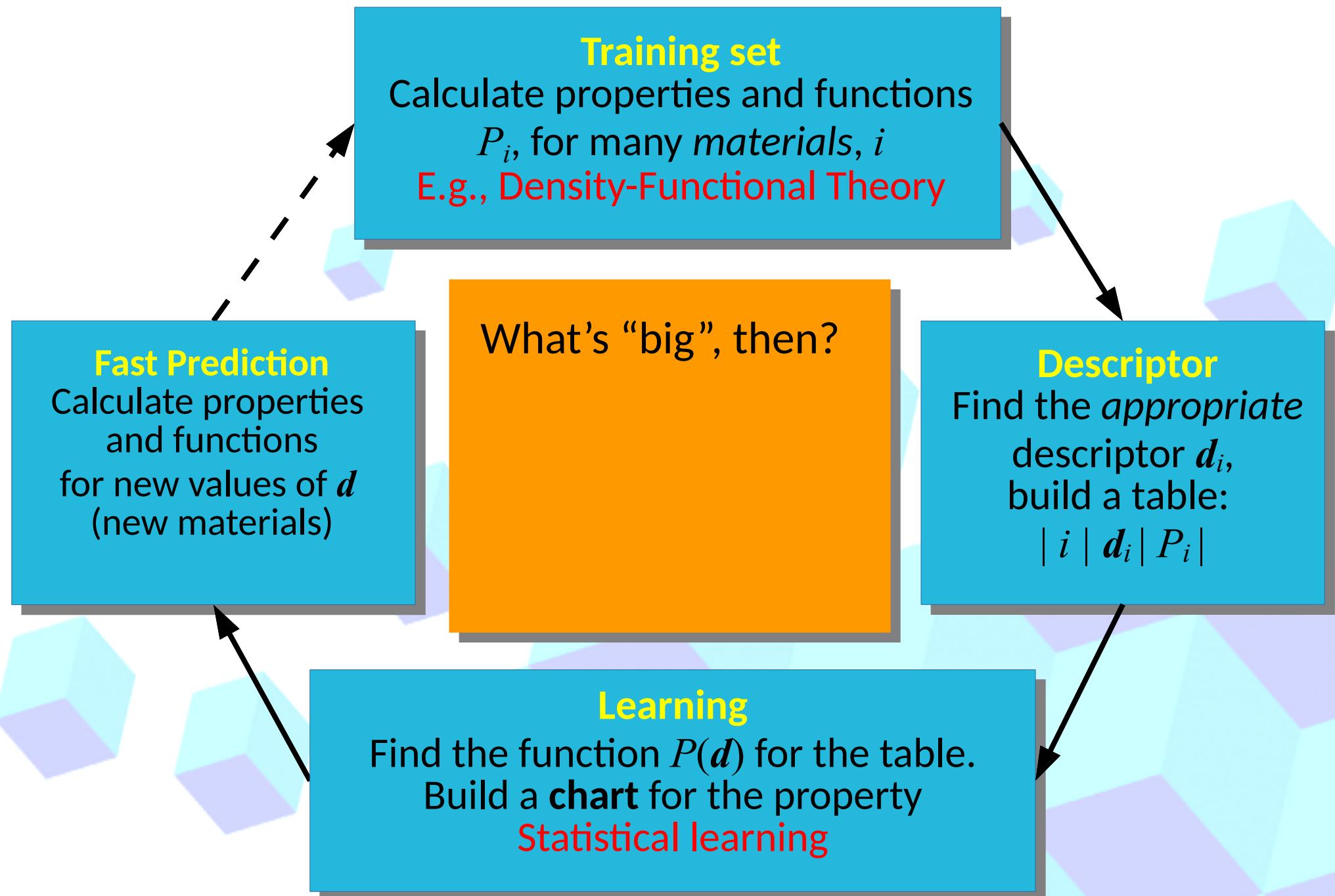
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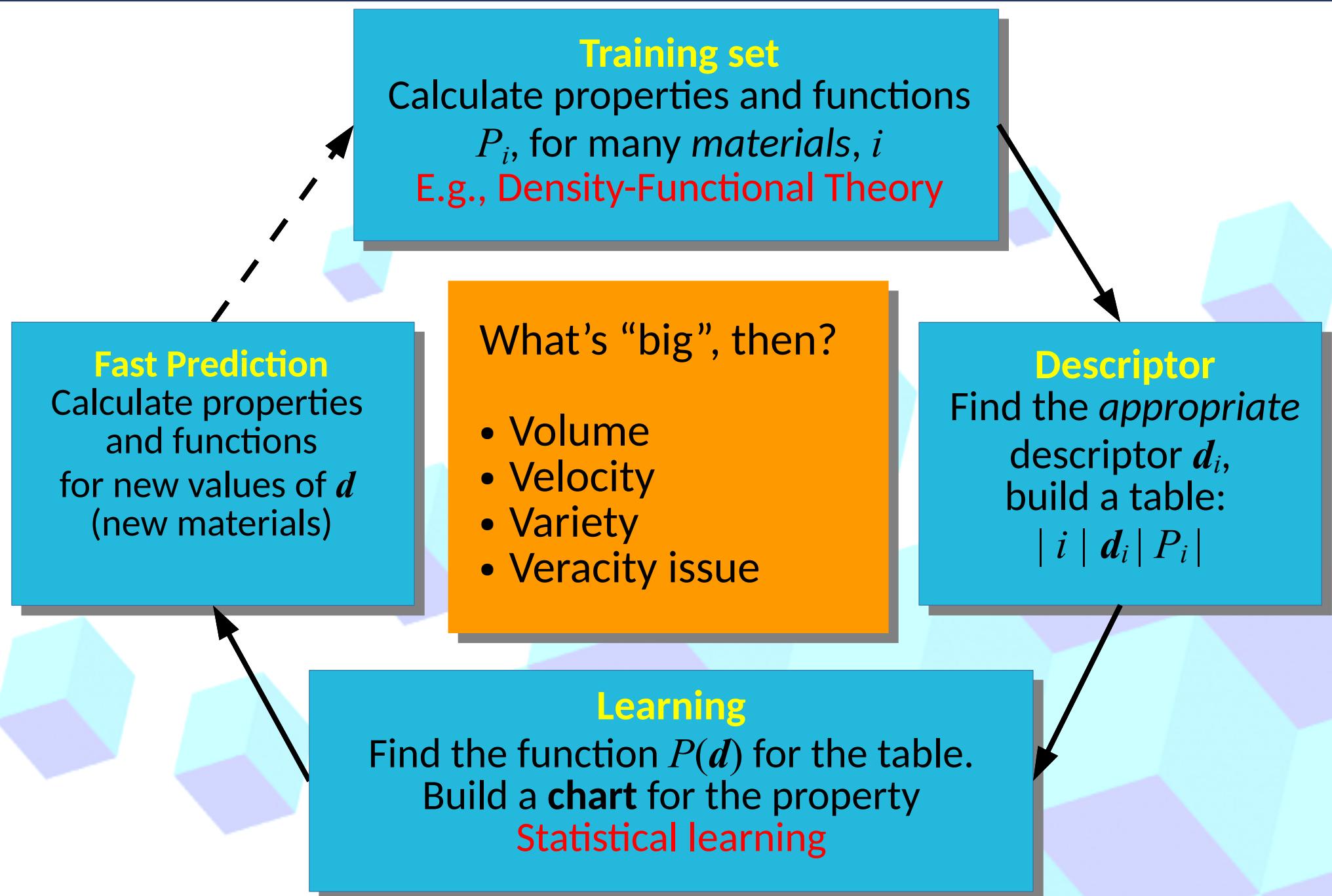
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Supervised big-data analysis: a flow chart



Supervised big-data analysis: a flow chart



Descriptor? Don't we know it from the start?

Training set

Calculate properties and functions

P_i , for many materials, i

E.g., Density-Functional Theory

$\{R_I, Z_I\} \rightarrow$ Hamiltonian

$\{R_I\} \rightarrow$ Geometry

- translational, rotational, permutational invariant

- coarse graining $\{R_I\}$?

$\{Z_I\} \rightarrow$ Chemistry

Descriptor

Find the appropriate descriptor d_i , build a table:

i	d_i	P_i
-----	-------	-------

Learning

Find the function $P(d)$ for the table.

Build a chart for the property

Statistical learning

Regression: Mathematical formulation

Figure of merit to be optimized:

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} \sum_{j=1}^N \left(P_j - \sum_{l=1}^M d_{j,l} c_l \right)^2 = \operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2$$

ℓ_2 norm

Ridge Regression: Mathematical formulation

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Regularization (prefer “lower complexity” in the solution)

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_2^2 \quad (\text{Linear}) \text{ ridge regression}$$

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Explicit solver:

$$\mathbf{c} = (\mathbf{D}^\top \mathbf{D} + \lambda \mathbf{I})^{-1} \mathbf{D}^\top \mathbf{P}$$

Alternative view, via Hilbert space representation theorem:

$$\mathbf{c} = \sum_j \alpha_j \mathbf{d}_j \quad \text{Sum over data points!}$$

Kernel Ridge Regression: Mathematical formulation

$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_2^2 \quad \Rightarrow \quad \mathbf{c} = (\mathbf{D}^\top \mathbf{D} + \lambda \mathbf{I})^{-1} \mathbf{D}^\top \mathbf{P}$$

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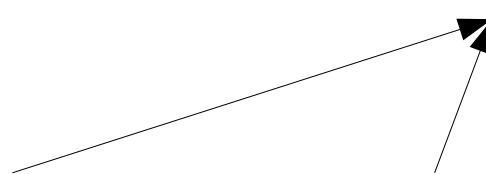


$$\mathbf{c} = \sum_j \alpha_j \mathbf{d}_j$$



$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} \| \mathbf{P} - \mathbf{K}\boldsymbol{\alpha} \|_2^2 + \lambda \boldsymbol{\alpha}^\top \mathbf{K} \boldsymbol{\alpha}$$

$$K_{ij} = \langle \mathbf{d}_i, \mathbf{d}_j \rangle \quad \text{Linear kernel}$$



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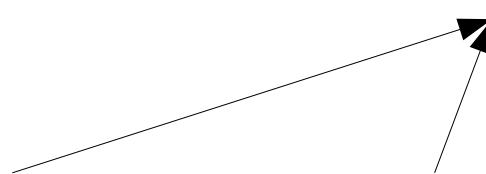


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Non-linear kernel

$$\mathbf{c} = \sum_j \alpha_j \Phi(\mathbf{d}_j)$$

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Kernel Ridge Regression: Mathematical formulation

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$$K_{ij} = \langle \mathbf{d}_i, \mathbf{d}_j \rangle \quad \text{Linear kernel}$$

$$K_{ij} = (\langle \mathbf{d}_i, \mathbf{d}_j \rangle + b)^n \quad \text{Polynomial kernel}$$

$$K_{ij} = \exp\left(\frac{\|\mathbf{d}_i - \mathbf{d}_j\|^2}{2\sigma^2}\right) \quad \text{Gaussian (radial basis function) kernel}$$

$$K_{ij} = \exp\left(\frac{\|\mathbf{d}_i - \mathbf{d}_j\|}{\sigma}\right) \quad \text{Laplacian kernel}$$

Kernel Ridge Regression: Mathematical formulation

Non-linear kernel

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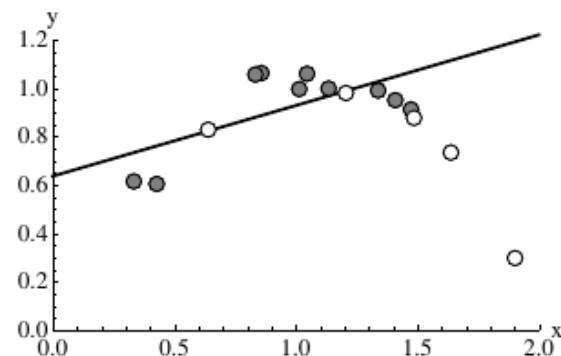
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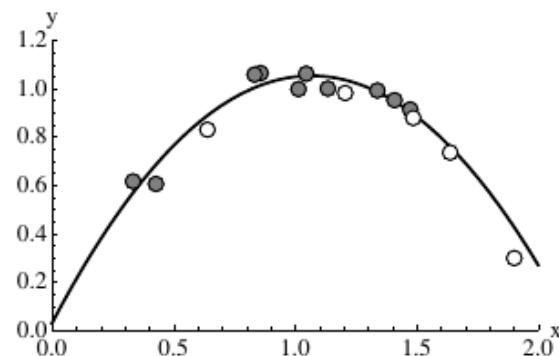
In all cases,
a kernel introduces a
similarity measure

Regularized regression in practice: beware of overfitting

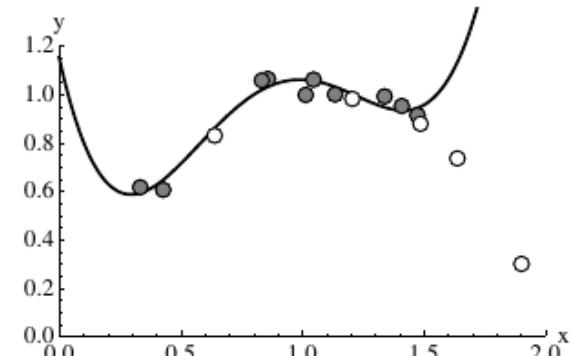
Underfitting



Fitting



Overfitting



Training/
validation
error

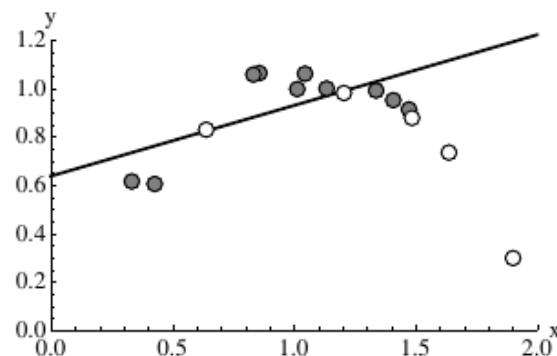
0.123 / 0.443

0.044 / 0.068

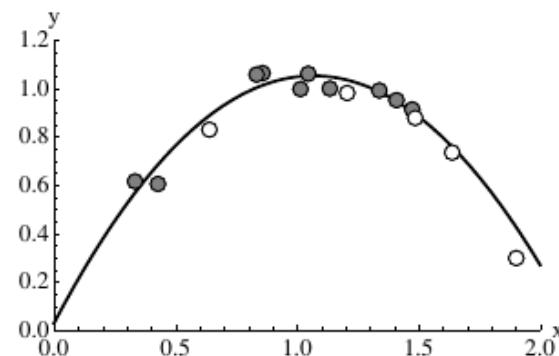
0.036 / 0.939

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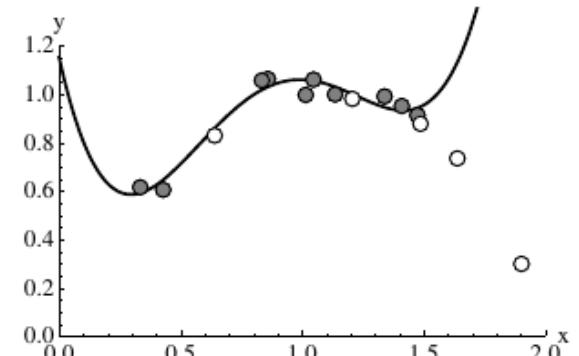
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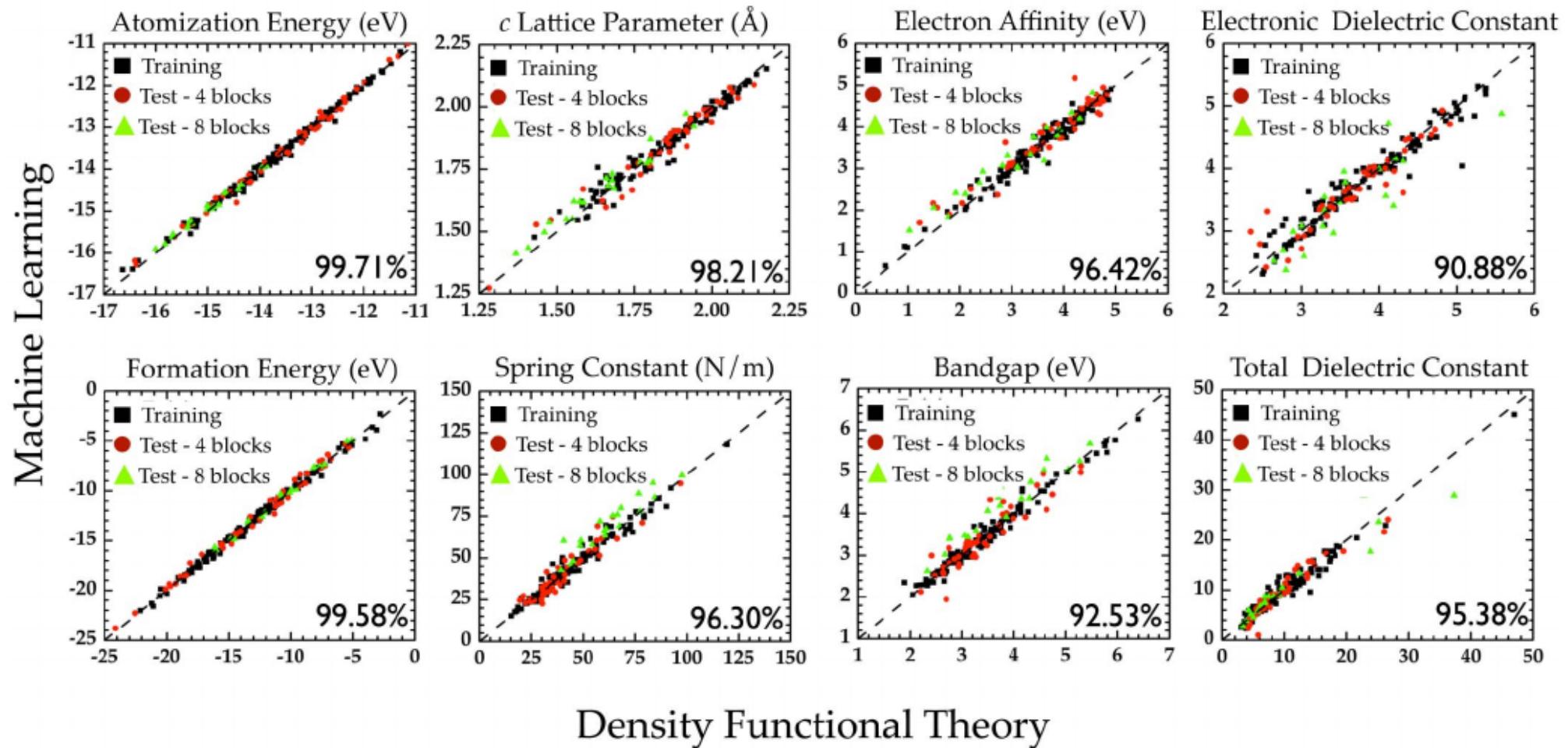
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KRR success stories: 1D polymers “eugenetics”

Data: 175 linear 4-blocks periodic polymers. 7 blocks: CH_2 , SiF_2 , SiCl_2 , GeF_2 , GeCl_2 , SnF_2 , SnCl_2 ,
Descriptor: 20 dimensions [# building blocks of type *i*, of *ii* pairs, of *iii* triplets]



KRR success stories: n -grams for kaggle

Research Prediction Competition

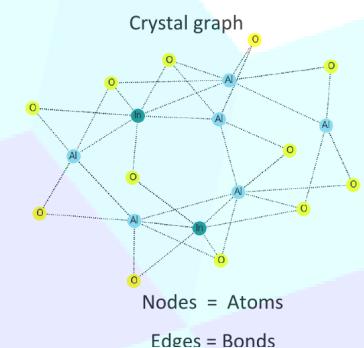
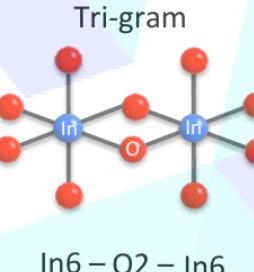
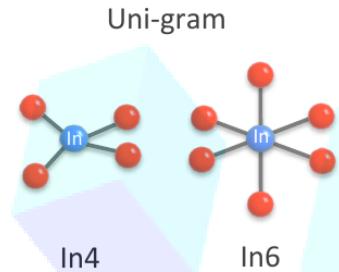
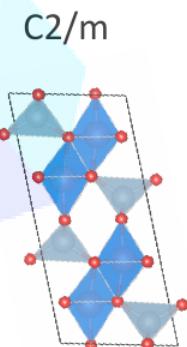
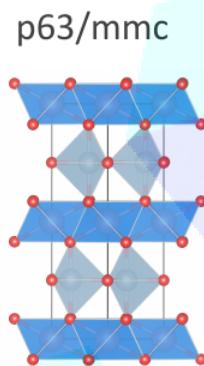
Nomad2018 Predicting Transparent Conductors

Predict the key properties of novel transparent semiconductors

883 teams · 12 days ago

€5,000 Prize Money

Overview Data Kernels Discussion Leaderboard Rules Team Host My Submissions Late Submission



Input features: count number of sequences of various lengths



Unigrams: 2 O3, 2 Ga4, 1 O2, 1 In5

Bigrams: 2 O3-Ga4, 2 Ga4-O2, 1 O3-In5

Compressed sensing: the quest for descriptors and predictive models

$$\arg \min_{\mathbf{c}} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

Compressed-sensing-based model identification:
Shares concepts with

Regularized regression. But: Massive sparsification.

Dimensionality reduction. But supervised, and yielding sparse,
“inspectable” descriptors

Feature/Basis-set selection/extraction. But: non-greedy solver.

Symbolic regression. But: deterministic solver.

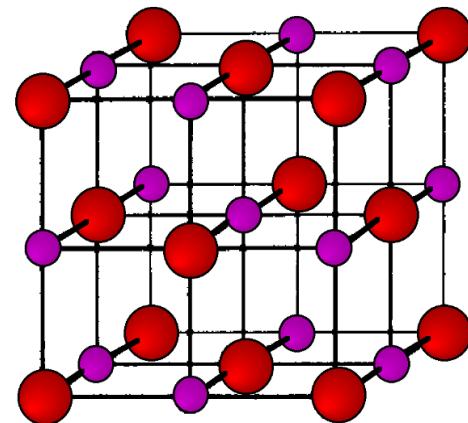
An example: predicting crystal structures from the composition

82 octet AB binary compounds

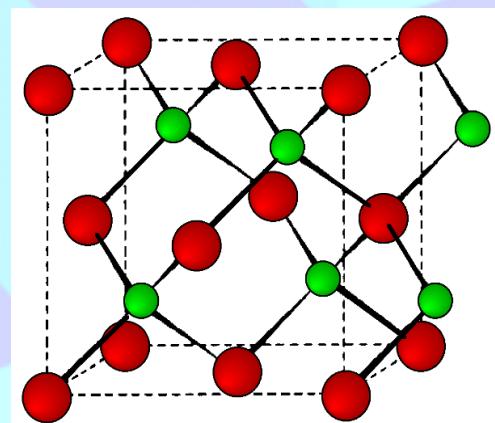
hydrogen 1 H	beryllium 4 Be
lithium 3 Li 6.941	magnesium 12 Mg 24.305
sodium 11 Na 22.990	calcium 20 Ca 40.078
potassium 19 K 39.098	strontium 38 Sr 87.62
rubidium 37 Rb 85.468	barium 56 Ba 137.33
cesium 55 Cs 132.91	lanthanum 57 La 138.91
francium 89 Fr 223.01	cerium 58 Ce 140.12
copper 29 Cu 63.546	praseodymium 59 Pr 140.91
zinc 30 Zn 65.39	neodymium 60 Nd 144.24
silver 47 Ag 107.87	promethium 61 Pm [145]
cadmium 48 Cd 130.90	samarium 62 Sm 150.36

** Actinide series

lanthanum 57 La 138.91	cerium 58 Ce 140.12	praseodymium 59 Pr 140.91	neodymium 60 Nd 144.24	promethium 61 Pm [145]	samarium 62 Sm 150.36	euroopium 63 Eu 151.96	gadolinium 64 Gd 157.25	terbium 65 Tb 158.93	dysprosium 66 Dy 162.50	holmium 67 Ho 164.93	erbium 68 Er 167.26	thulium 69 Tm 168.93	yterbium 70 Yb 173.04
actinium 89 Ac [227]	thorium 90 Th [232.04]	protactinium 91 Pa [231.04]	uranium 92 U [238.03]	neptunium 93 Np [237]	plutonium 94 Pu [244]	americium 95 Am [243]	curium 96 Cm [247]	berkelium 97 Bk [247]	einsteinium 98 Cf [251]	californium 99 Es [252]	fermium 100 Fm [257]	mendelevium 101 Md [258]	nobelium 102 No [259]



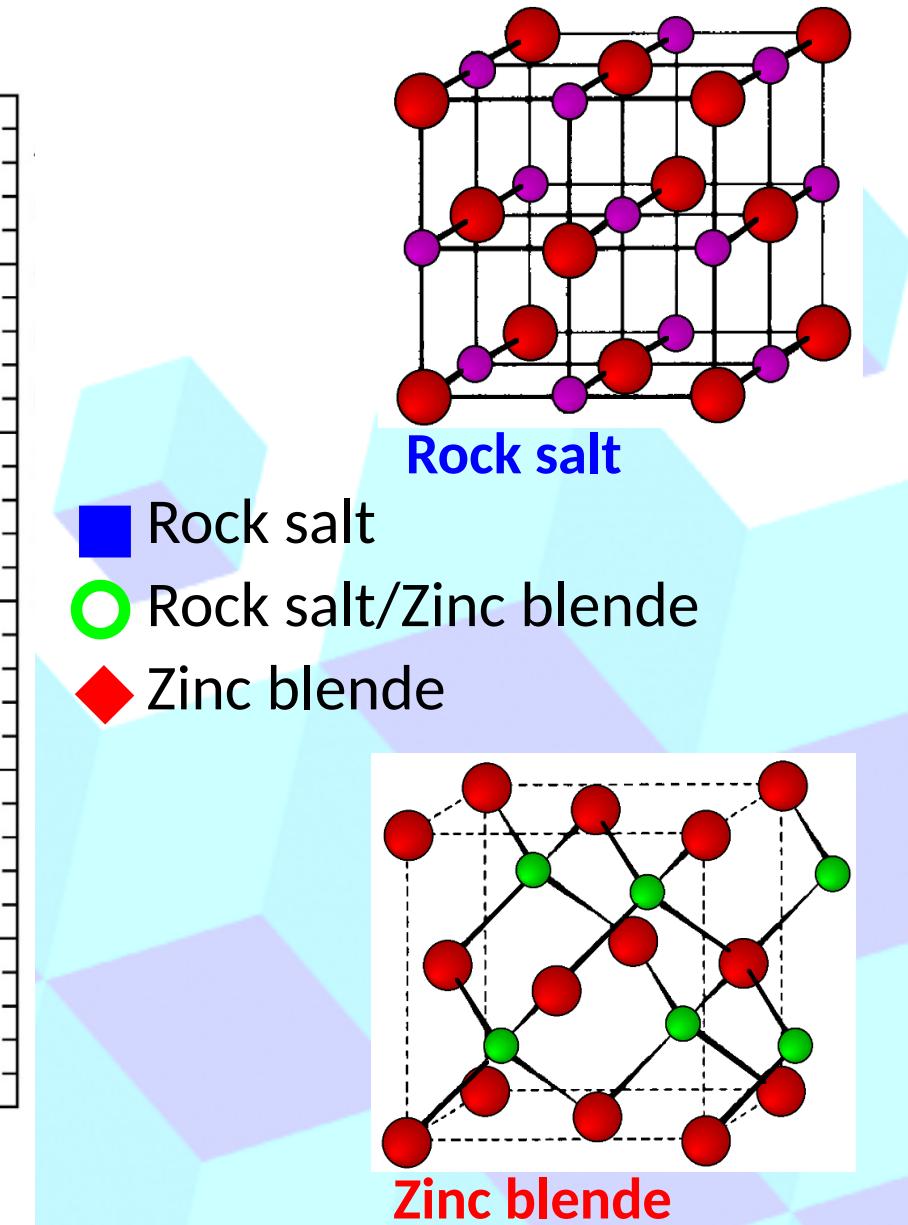
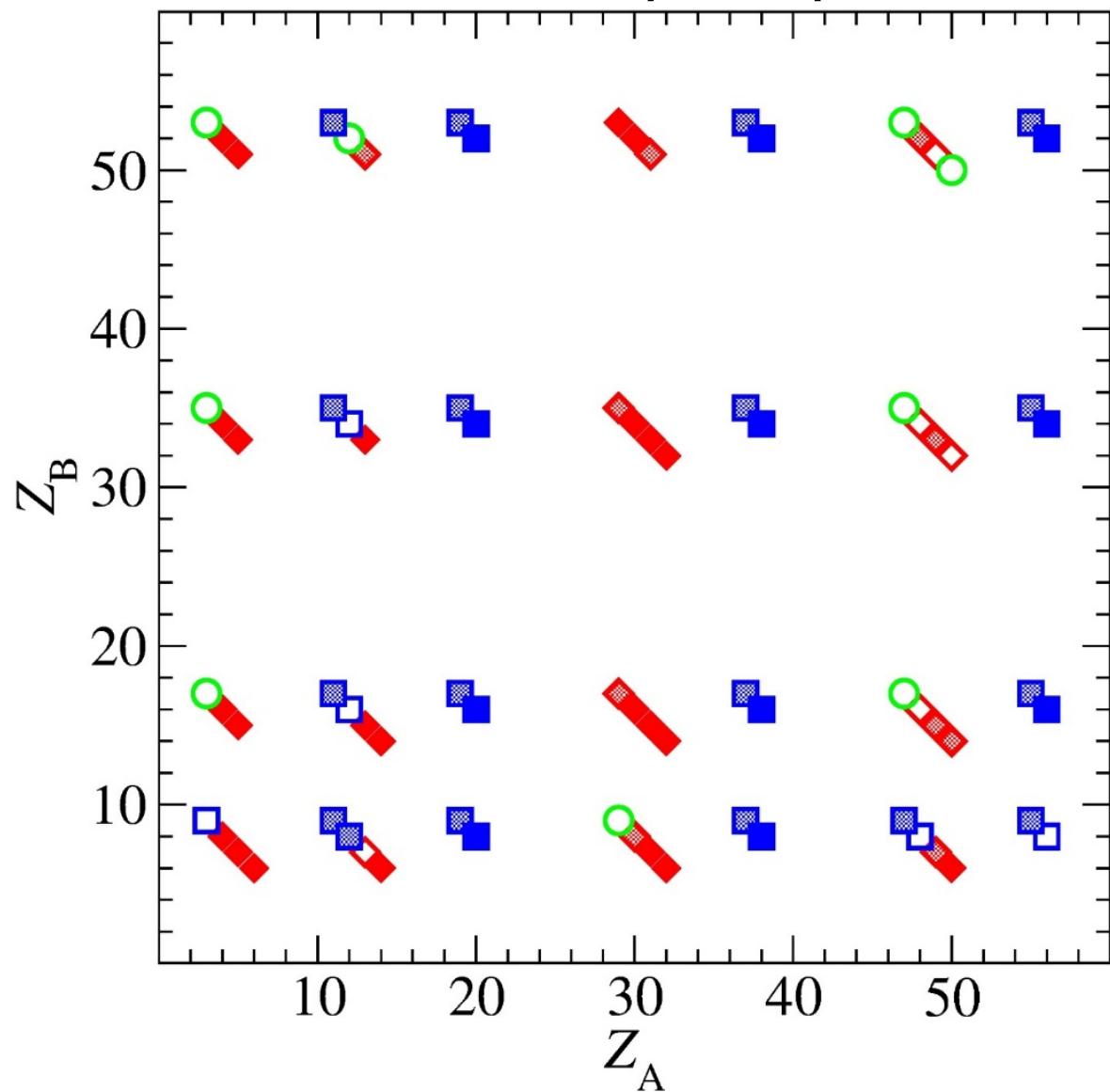
Rock salt



Zinc blende

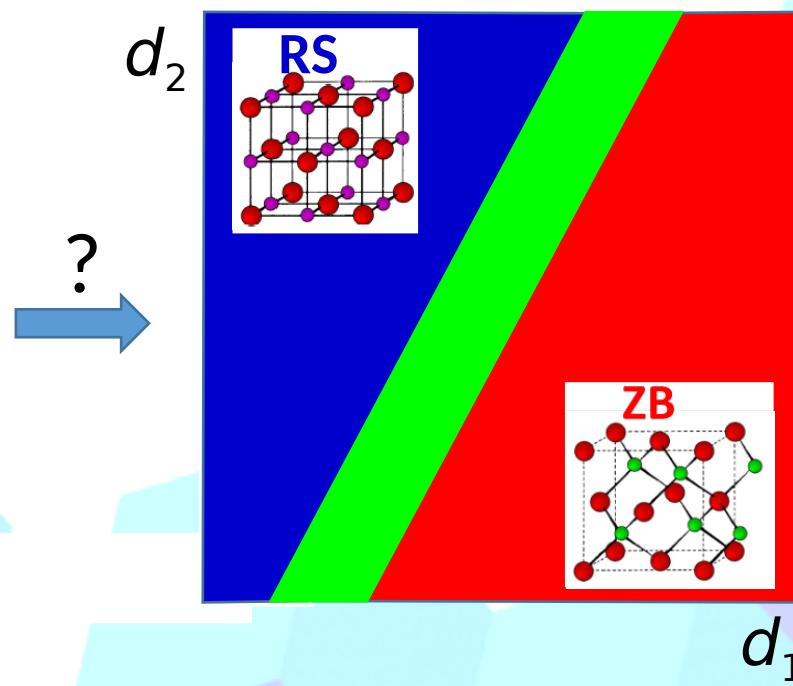
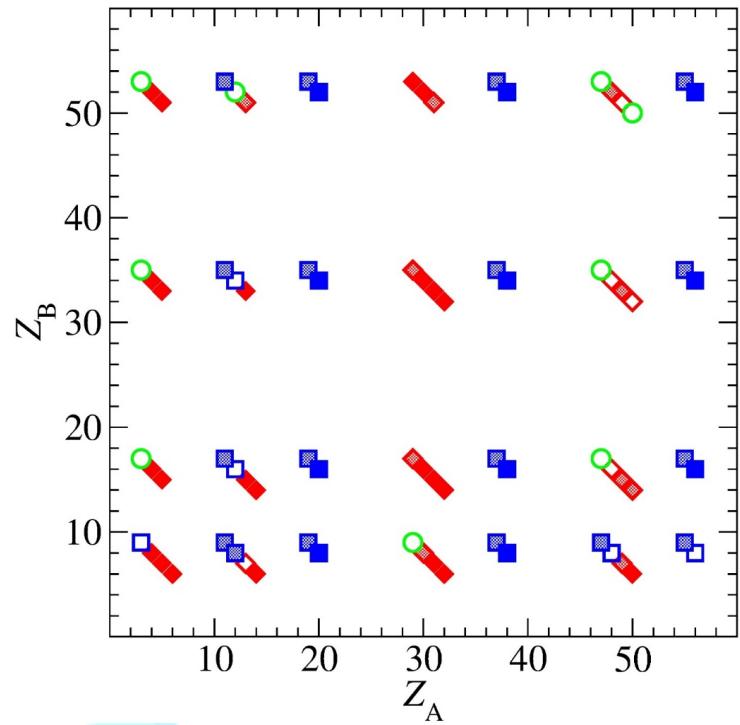
An example: predicting crystal structures from the composition

82 octet AB binary compounds



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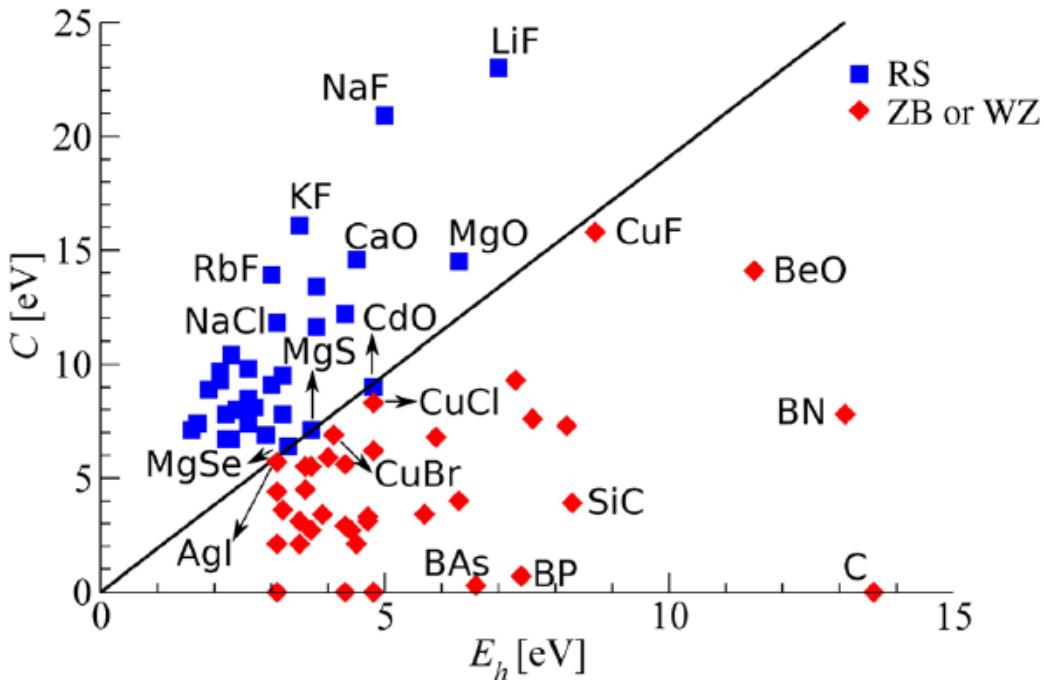


- Rock salt
- Rock salt/Zinc blende
- ◆ Zinc blende

- J. A. van Vechten, Phys. Rev. 182, 891 (1969).
J. C. Phillips, Rev. Mod. Phys. 42, 317 (1970).
J. John and A.N. Bloch, Phys. Rev. Lett. 33, 1095 (1974)
J. R. Chelikowsky and J. C. Phillips, Phys. Rev. B 33, 2453 (1978)
A. Zunger, Phys. Rev. B 22, 5839 (1980).
D. G. Pettifor, Solid State Commun. 51, 31 (1984).
Y. Saad, D. Gao, T. Ngo, S. Bobbitt, J. R. Chelikowsky, and W. Andreoni, Phys. Rev. B 85, 104104 (2012).

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The descriptor proposed by Phillips and van Vechten in 1969-70 depends on:
- lattice parameter
- electrical conductivity

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J. C. Phillips, Rev. Mod. Phys. 42, 317 (1970).

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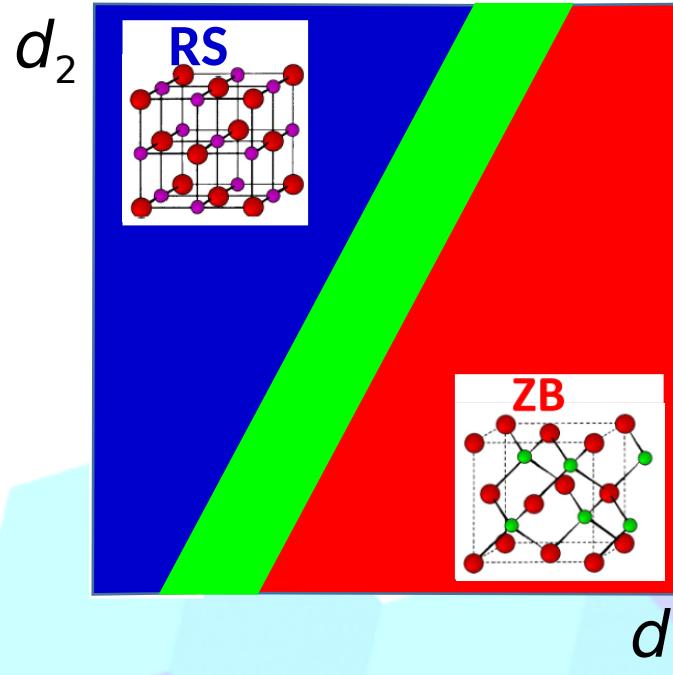
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82 octet AB binary compounds

Ansatz: atomic features

- HOMO
- LUMO
- Ionization Potential
- Electron Affinity
- Radius of valence s orbital
- Radius of valence p orbital
- Radius of valence d orbital
- ... ?

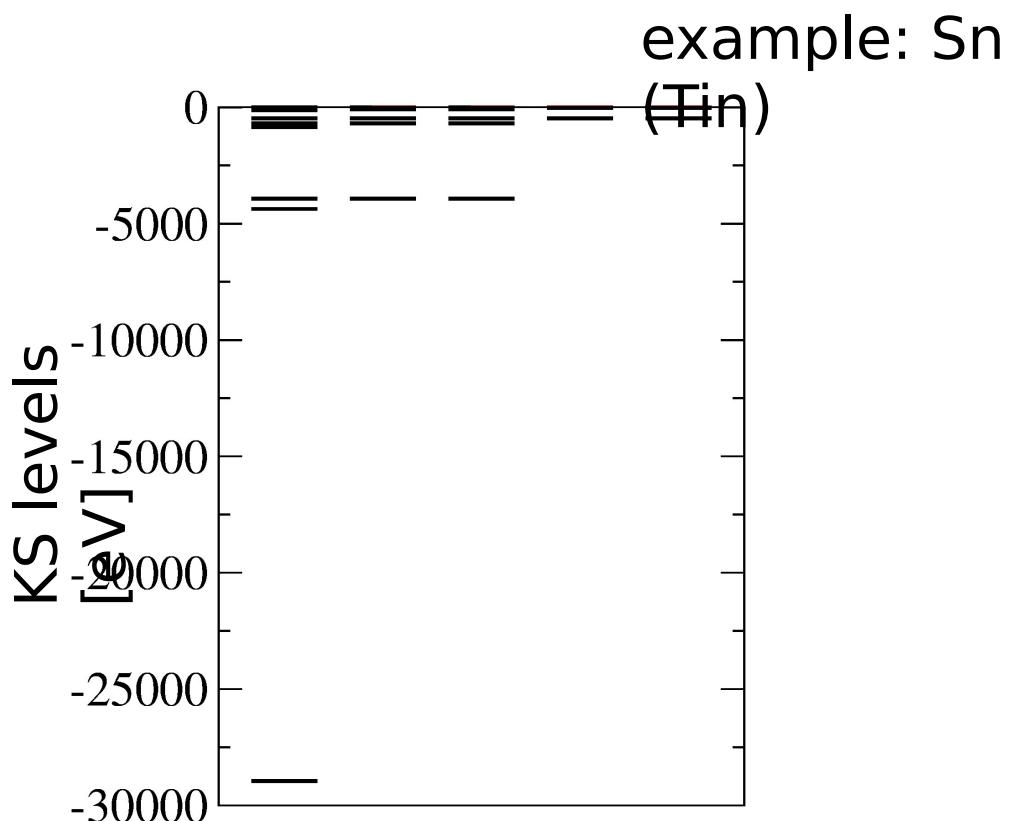


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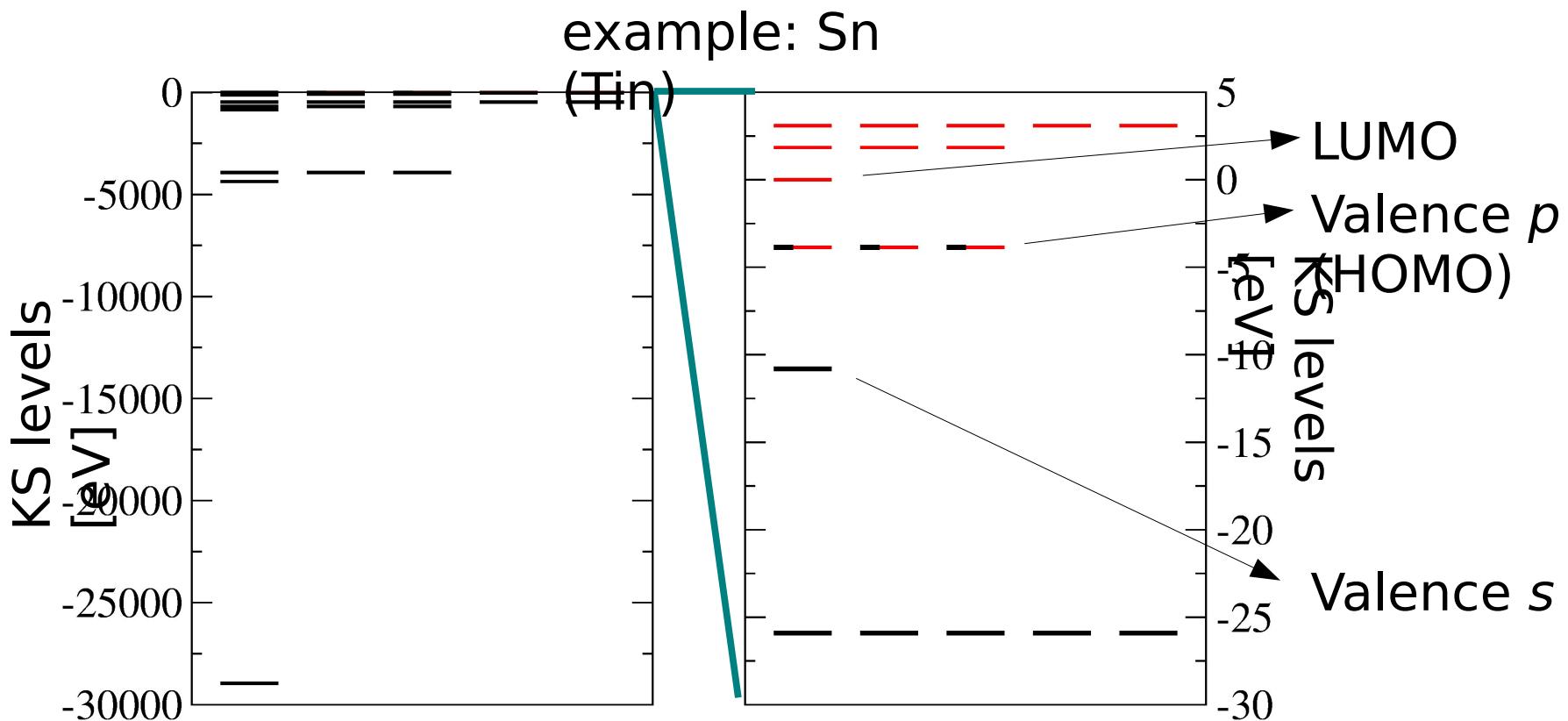
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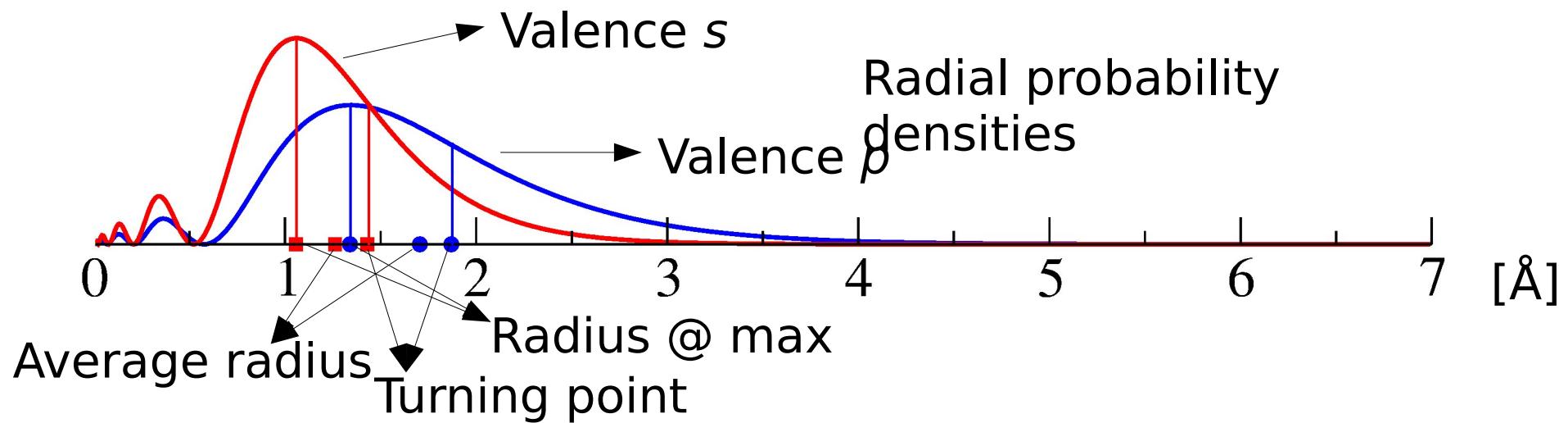
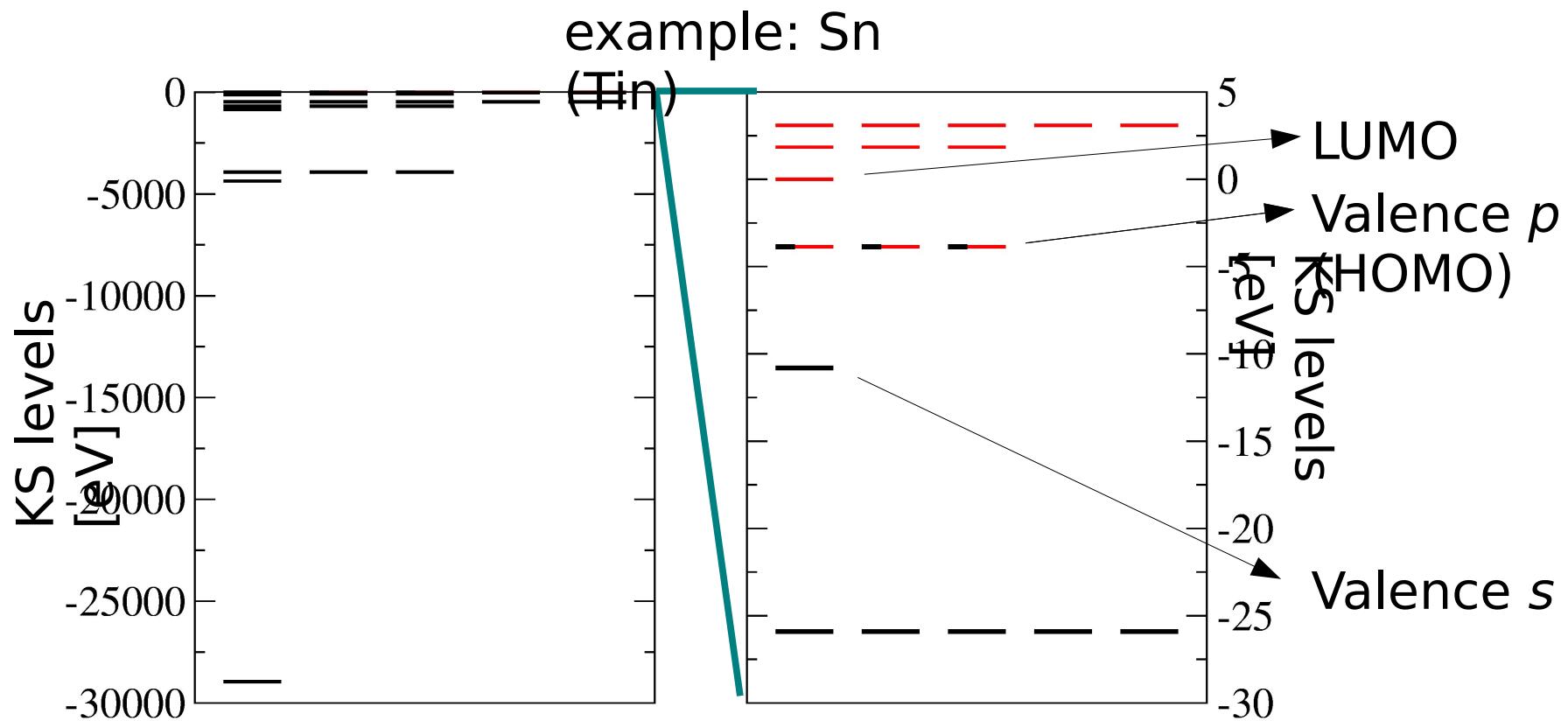
Primary (atomic) features



Primary (atomic) features



Primary (atomic) features

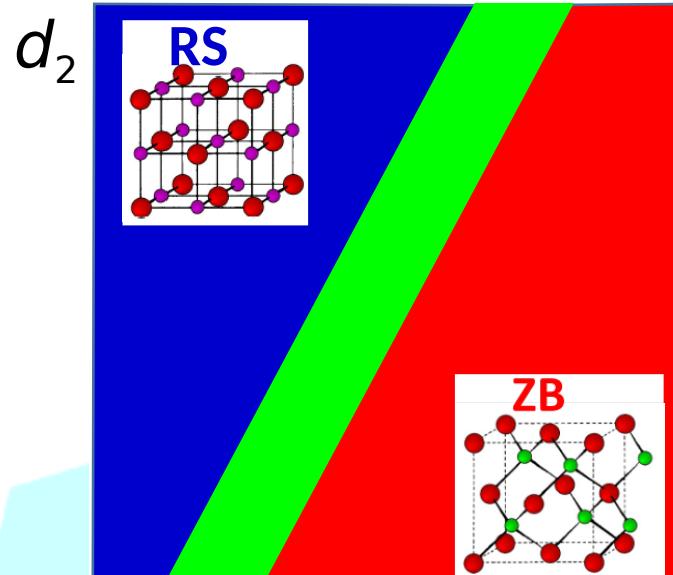


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Compressed sensing

Aim: finding descriptors and learning predictive models

Ansatz:

$$P = c_1 \mathbf{d}_1 + c_2 \mathbf{d}_2 + \dots + c_n \mathbf{d}_n$$

Where

P is the property of interest

$\mathbf{d}_1, \dots, \mathbf{d}_n$ are candidate features, i.e., nonlinear functions of primary features (EA, IP, ...)

c_1, \dots, c_n are unknown coefficients, with the extra constraint that these (nonzero) coefficients should be as few as possible.

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With a foreword on
dimensionality reduction

Linear dimensionality reduction: Principal components

Pearson, K. "On Lines and Planes of Closest Fit to Systems of Points in Space".
Philosophical Magazine 2, 559 ([1901](#))

Linear dimensionality reduction: Principal components

Pearson, K. "On Lines and Planes of Closest Fit to Systems of Points in Space".
Philosophical Magazine 2, 559 (1901)

Orthonormal transformation of coordinates, converting a set of (possibly) linearly correlated coordinates into a new set of linearly uncorrelated (called principal or normal) components, such that the first component has the largest variance and each subsequent has the largest variance constrained to being orthogonal to all the preceding components

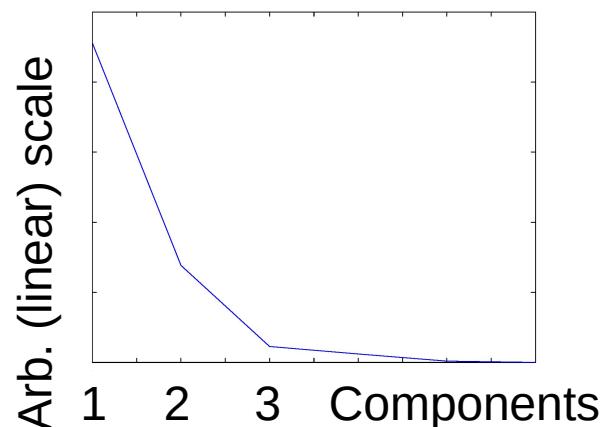
Linear dimensionality reduction: Principal components

Ansatz: atomic features

- Valence number Z_v
- Energy of valence s orbital E_s
- Energy of valence p orbital E_p
- Radius of valence s orbital r_s
- Radius of valence p orbital r_p

$r_s, r_p, E_s/\sqrt{Z_v}, E_p/\sqrt{Z_v}$,
for A and B atoms

linearly uncorrelated (called principal or normal) components, such that the first component has the largest variance and each subsequent has the largest variance constrained to being orthogonal to all the preceding components



Linear dimensionality reduction: Principal components

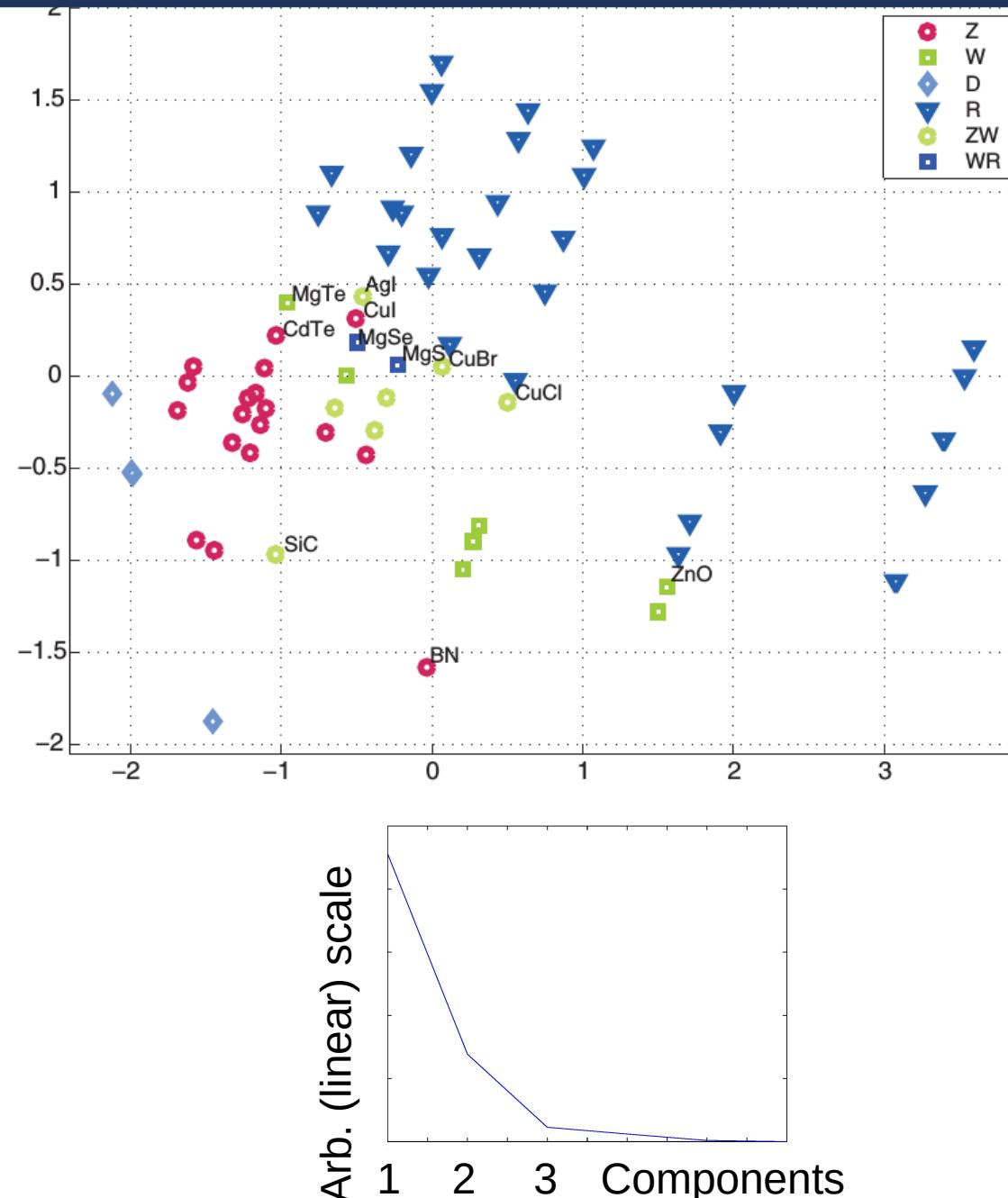
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Z_v
 E_s
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 r_s
 r_p

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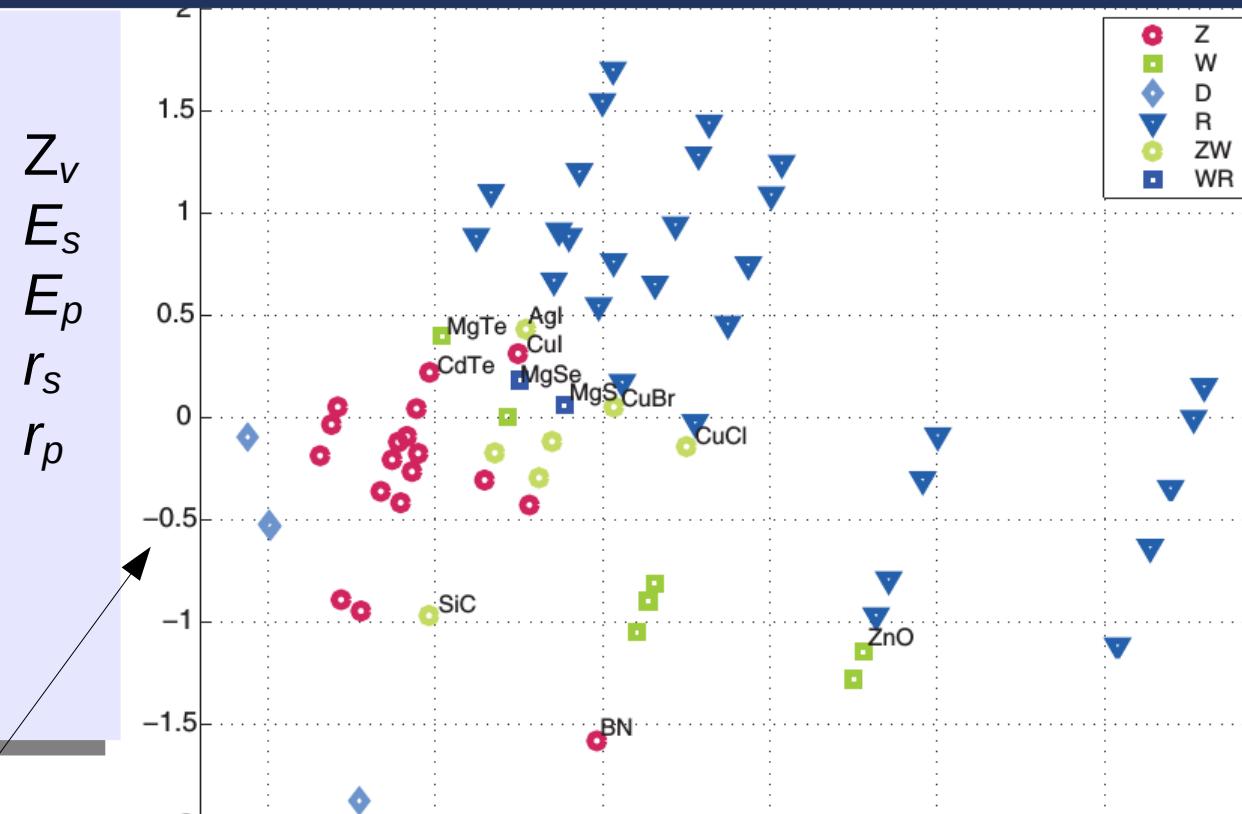


Linear dimensionality reduction: Principal components

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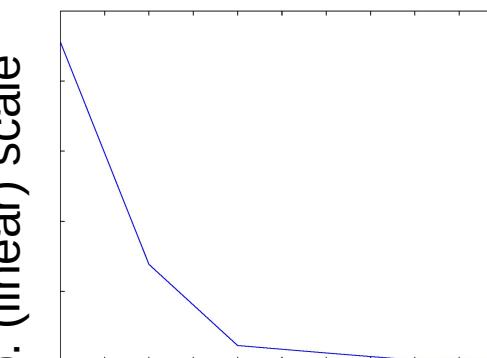
- Valence number
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- Radius of valence s orbital
- Radius of valence p orbital

$r_s, r_p, E_s/\sqrt{Z_v}, E_p/\sqrt{z_v}$,
for A and B atoms



What's on the axes?

Linear combination
of (possibly all) the
initial dimensions

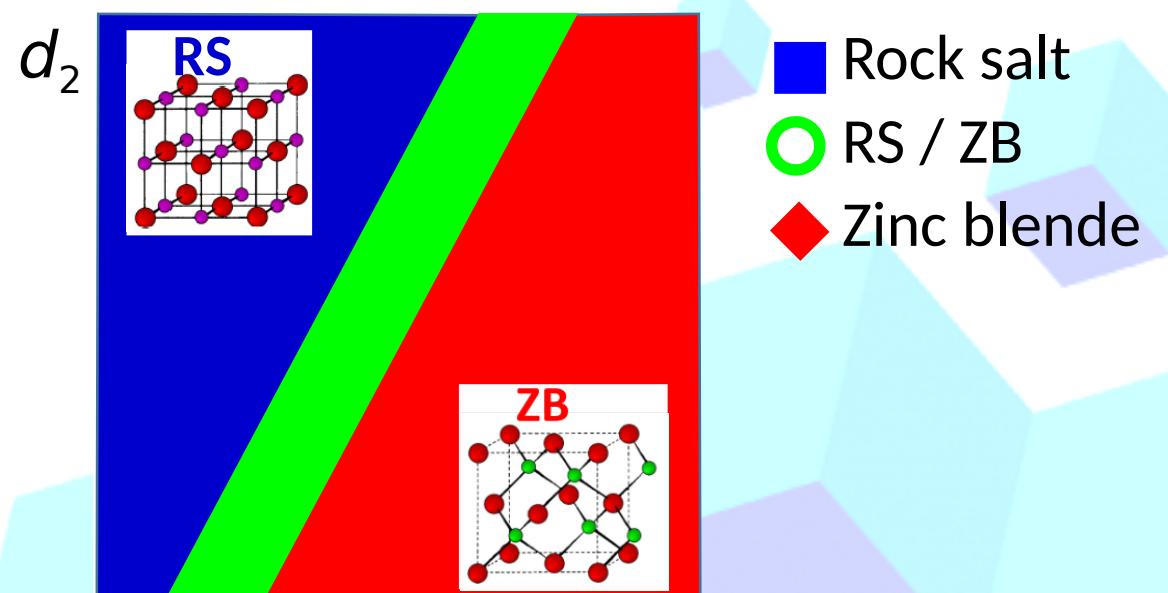


Compressed sensing: the quest for descriptors and predictive models

82 octet AB binary compounds

Ansatz: atomic features

- HOMO
- LUMO
- Ionization Potential
- Electron Affinity
- Radius of valence *s* orbital
- Radius of valence *p* orbital
- Radius of valence *d* orbital
- Thousands to billions of non-linear functions of the above



$$P = c_1 \mathbf{d}_1 + c_2 \mathbf{d}_2 + \dots + c_n \mathbf{d}_n$$

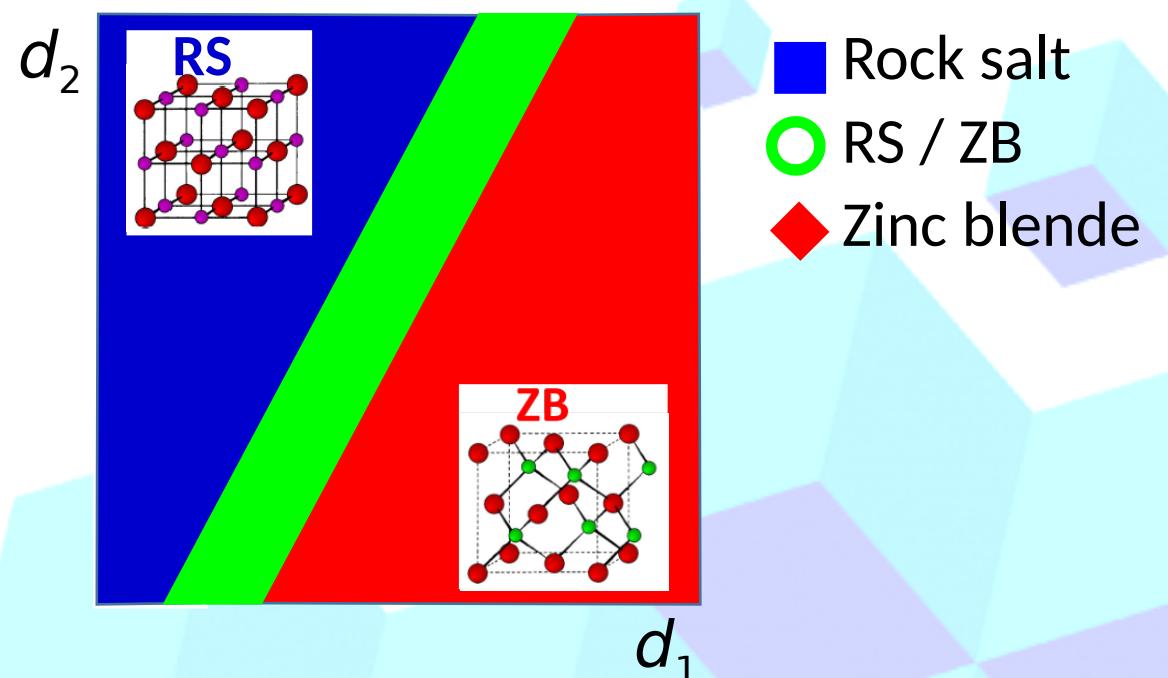
$E(\text{Rock salt}) - E(\text{Zinc blende})$

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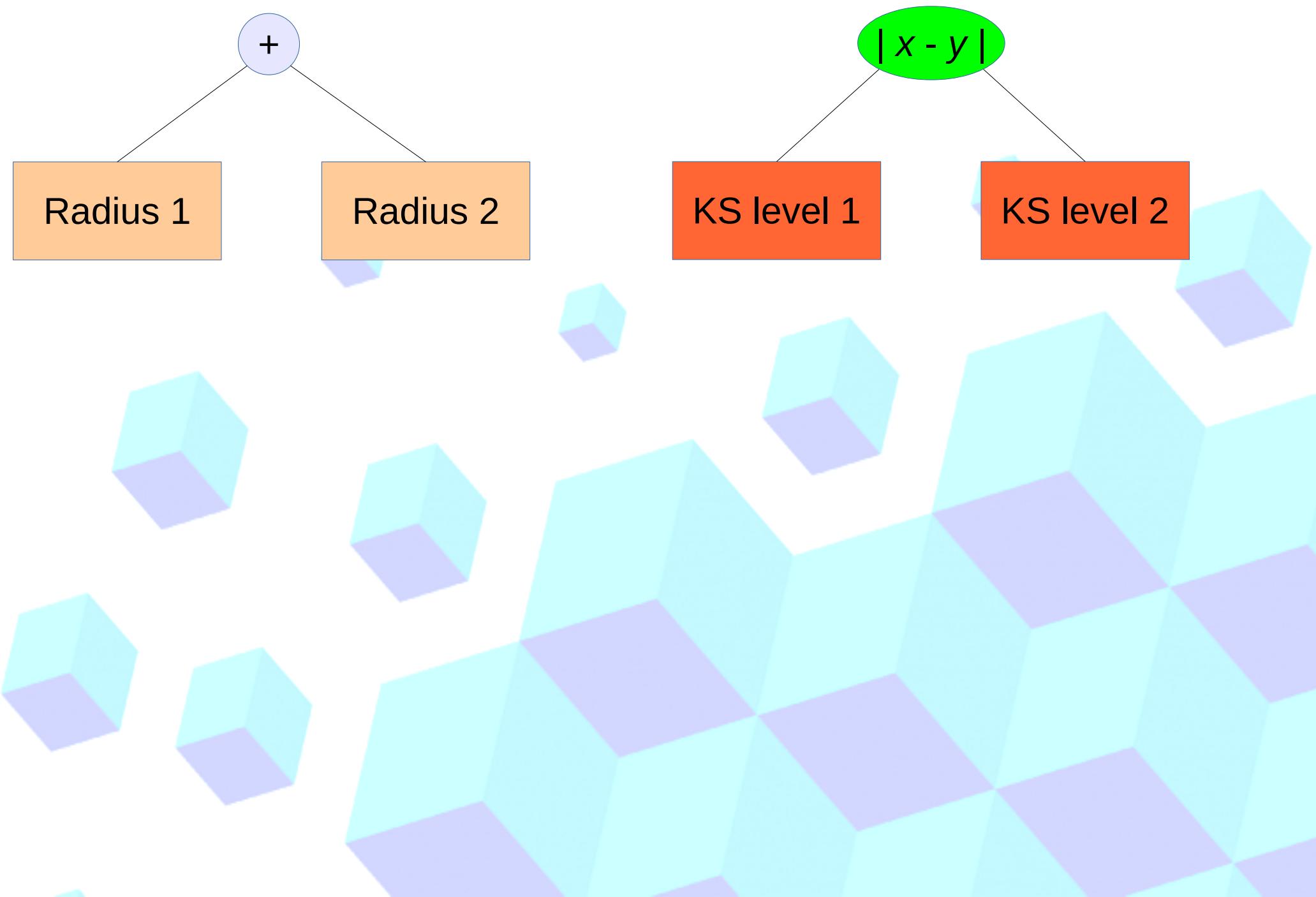


Symbolic Regression

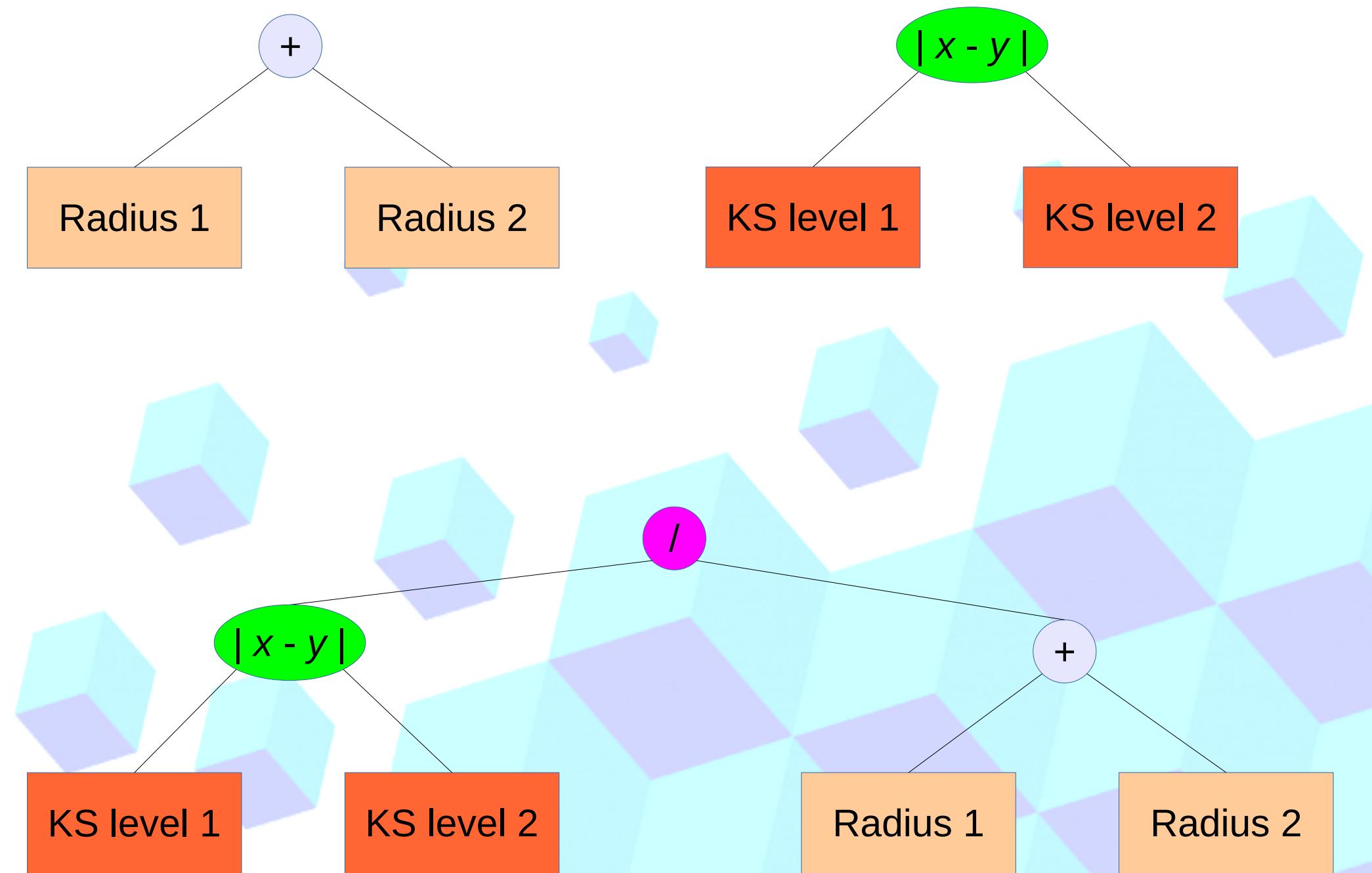
$E(\text{Rock salt}) - E(\text{Zinc blende})$

$$P = c_1 d_1 + c_2 d_2 + \dots + c_n d_n$$

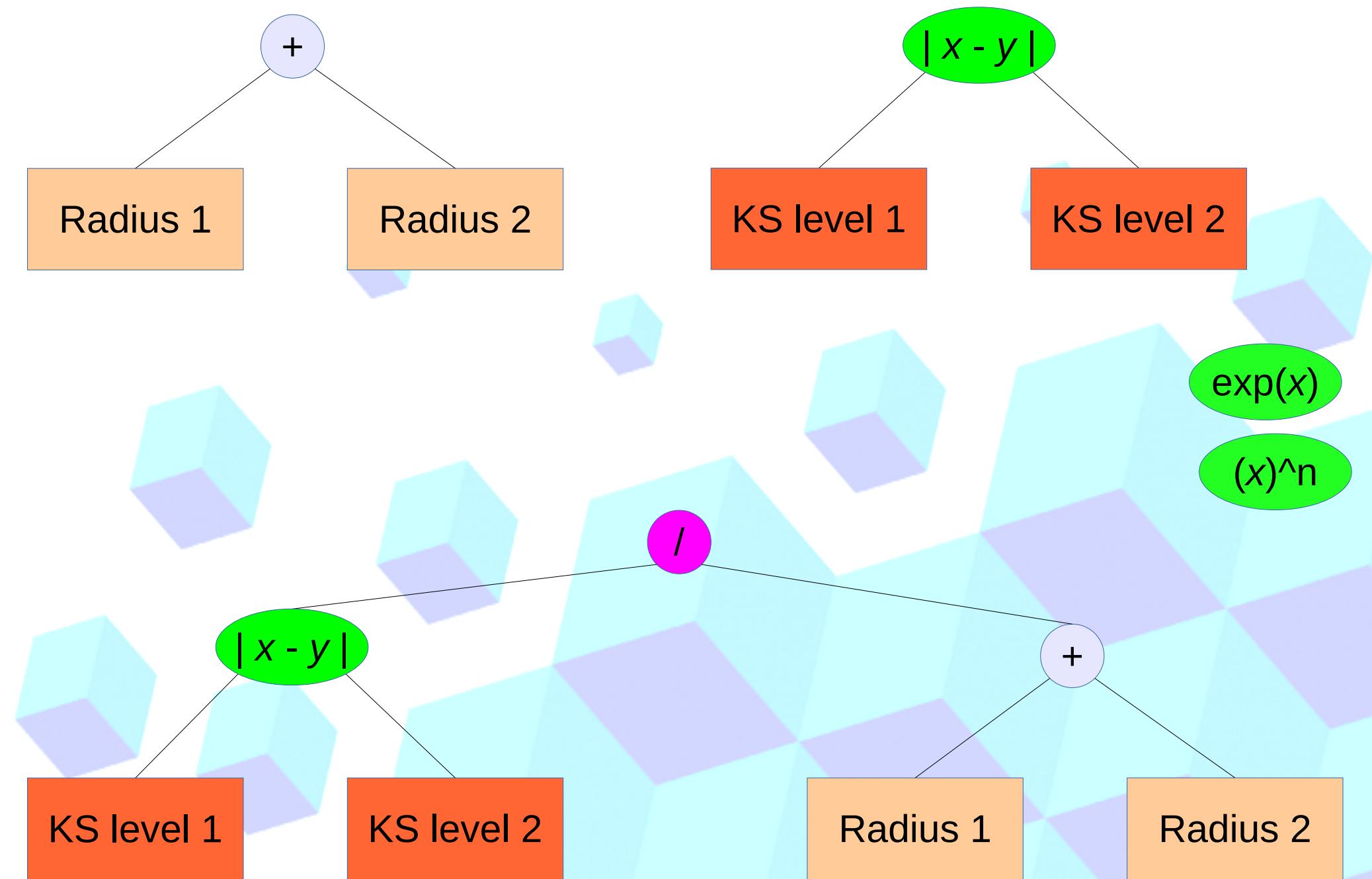
Systematic construction of the feature space



Systematic construction of the feature space



Systematic construction of the feature space



Systematic construction of the feature space: EUREQA

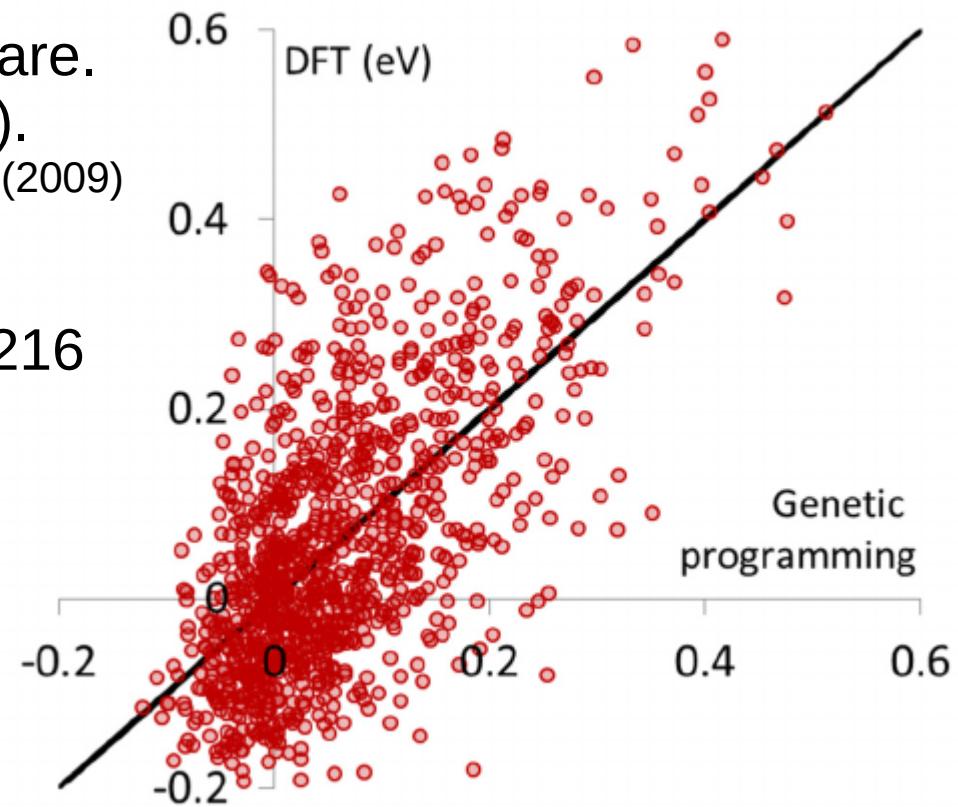
EUREQA: genetic programming software.
Global optimization (genetic algorithm).

Schmidt M., Lipson H., Science, Vol. 324, No. 5923, (2009)

T. Müller et al. PRB **89** 115202 (2014):
Data: ~1000 amorphous structures of 216
Si atoms (saturated)

Property: hole trap depth

$$\frac{\min(1.66355, a) \max(5.37551, c) - f - bd}{g} - h \max(3.42929, e),$$



Descriptor (candidates: 242)

- a The largest distance between a H atom and its nearest Si neighbor
- b The shortest distance between a Si atom and its sixth-nearest Si neighbor
- c The maximum bond valence sum on a Si atom
- d The smallest value for the fifth-smallest relative bond length around a Si atom
- e The fourth-shortest distance between a Si atom and its eighth-nearest neighbor
- f The second-shortest distance between a Si atom and its fifth-nearest neighbor
- g The third-shortest distance between a Si atom and its sixth-nearest neighbor
- h The H-Si nearest-neighbor distance for the hydrogen atom with the fourth-smallest difference between the distances to the two Si atoms nearest to a H atom

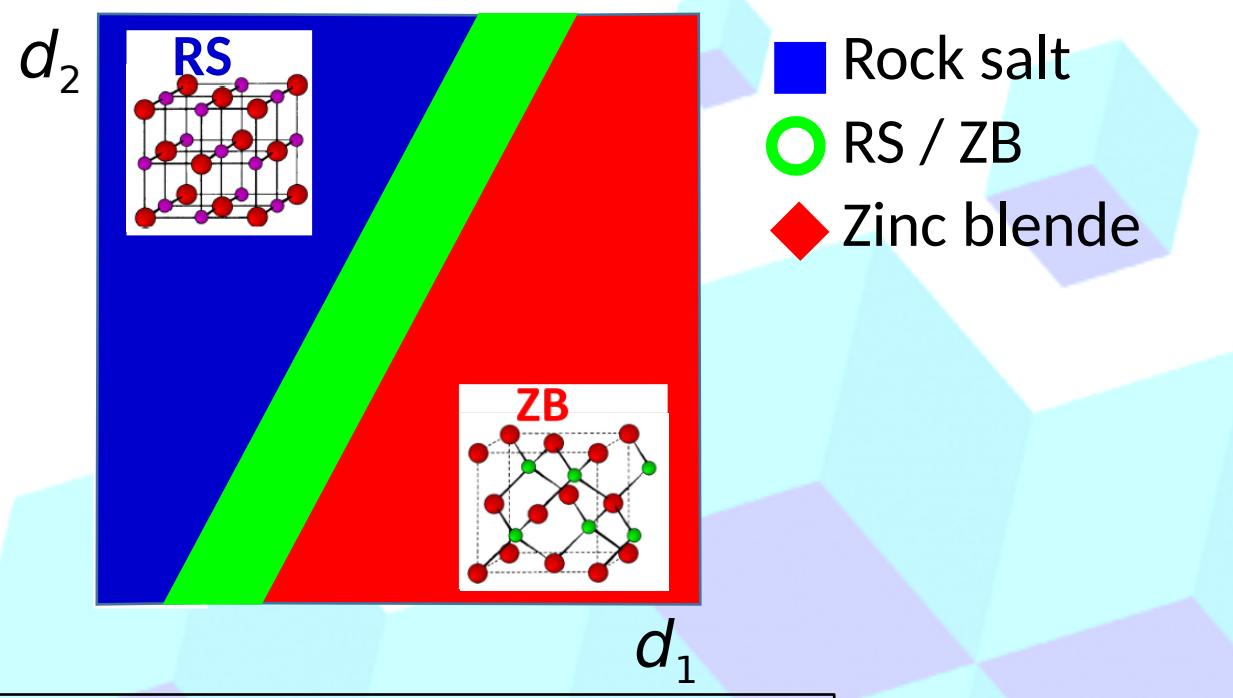
Building block	
Constant value	Exponential
Input variable	Natural logarithm
Addition	Power
Subtraction	Square root
Multiplication	Logistic function
Division	Minimum
Negation	Maximum
	Absolute value

Compressed sensing: the quest for descriptors and predictive models

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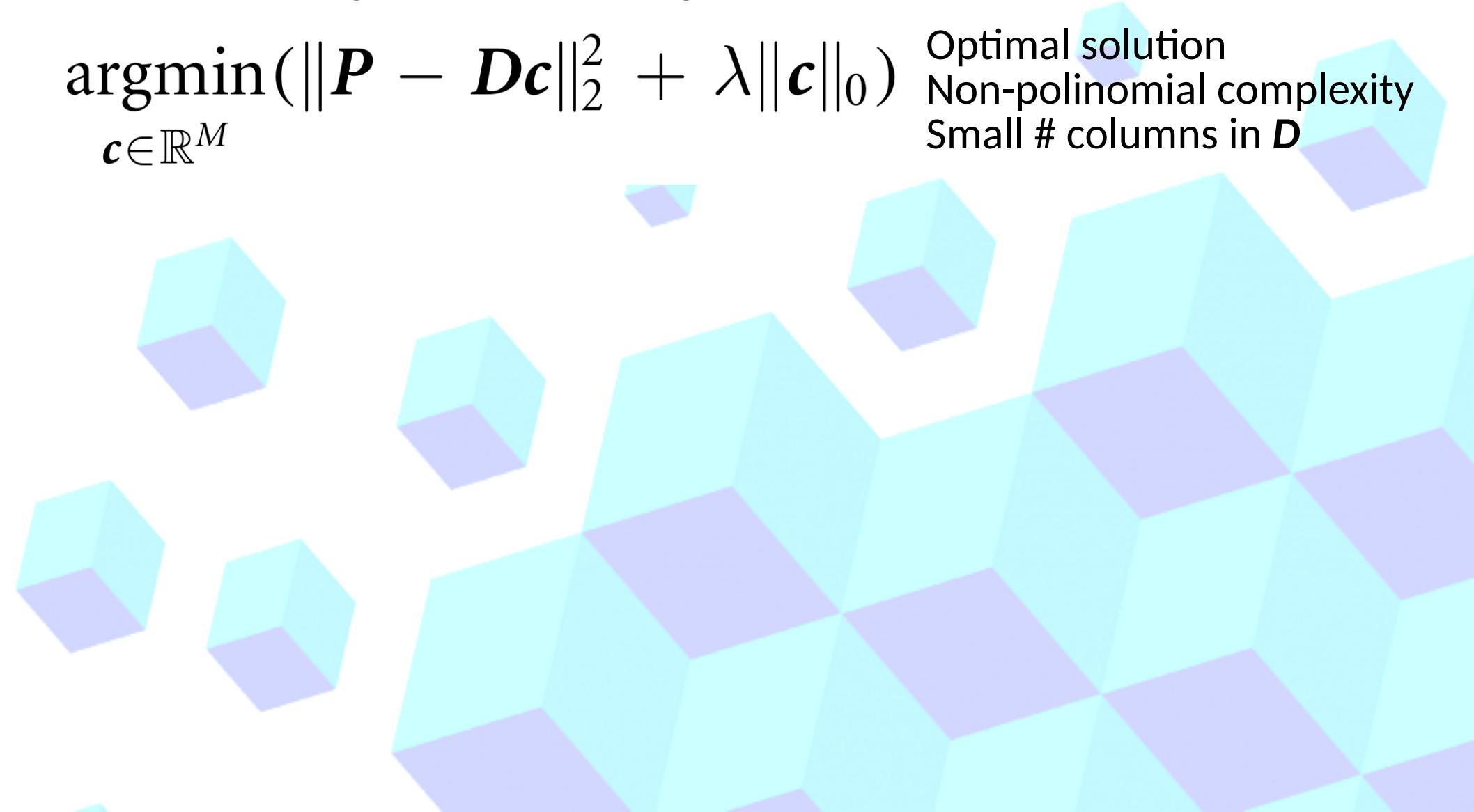
$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} \|P - D\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0$$

Compressed sensing: the quest for descriptors and predictive models

Ideal method: regression with ℓ_0 regularization

$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

Optimal solution
Non-polynomial complexity
Small # columns in \mathbf{D}



Compressed sensing: the quest for descriptors and predictive models

Ideal method: regression with ℓ_0 regularization

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Optimal solution
Non-polynomial complexity
Small # columns in \mathbf{D}

$\|\mathbf{c}\|_0$ # of nonzero elements of \mathbf{c}
 $\|\mathbf{c}\|_2$ Euclidean. Square root of sum of squares of the elements of \mathbf{c})

Compressed sensing: the quest for descriptors and predictive models

Ideal method: regression with ℓ_0 regularization

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

Optimal solution
Non-polynomial complexity
Small # columns in \mathbf{D}

$\|\mathbf{c}\|_0$ # of nonzero elements of \mathbf{c}
 $\|\mathbf{c}\|_2$ Euclidean. Square root of sum of squares of the elements of \mathbf{c})

For matrices D with uncorrelated columns: LASSO

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_1$$

(Possibly) optimal solution
Convex optimization
Moderate # columns in \mathbf{D}

$\|\mathbf{c}\|_1$ “Manhattan”. Sum of absolute values of the elements of \mathbf{c}

Compressed sensing in materials science

PRL 113, 185501 (2014)

PHYSICAL REVIEW LETTERS

week ending
31 OCTOBER 2014

Lattice Anharmonicity and Thermal Conductivity from Compressive Sensing of First-Principles Calculations

Fei Zhou (周非)

Physical and Life Sciences Directorate, Lawrence Livermore National Laboratory, Livermore, California 94550, USA

Weston Nielson, Yi Xia, and Vidvuds Ozoliņš

Department of Materials Science and Engineering, University of California, Los Angeles, California 90095-1595, USA

(Received 22 April 2014; published 27 October 2014)

Compressed modes for variational problems in mathematics and physics

Vidvuds Ozoliņš^{a,1}, Rongjie Lai^{b,1}, Russel Caflisch^{c,1}, and Stanley Osher^{a,1,2}

Departments of ^aMaterials Science and Engineering, and ^cMathematics, University of California, Los Angeles, CA 90095-1555; and ^bDepartment of Mathematics, University of California, Irvine, CA 92697-3875

Contributed by Stanley Osher, October 8, 2013 (sent for review September 3, 2013)

PHYSICAL REVIEW B 87, 035125 (2013)

Compressive sensing as a paradigm for building physics models

Lance J. Nelson and Gus L. W. Hart

Department of Physics and Astronomy, Brigham Young University, Provo, Utah 84602, USA

Fei Zhou (周非) and Vidvuds Ozoliņš*

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(Received 26 June 2012; revised manuscript received 26 September 2012; published 18 January 2013)



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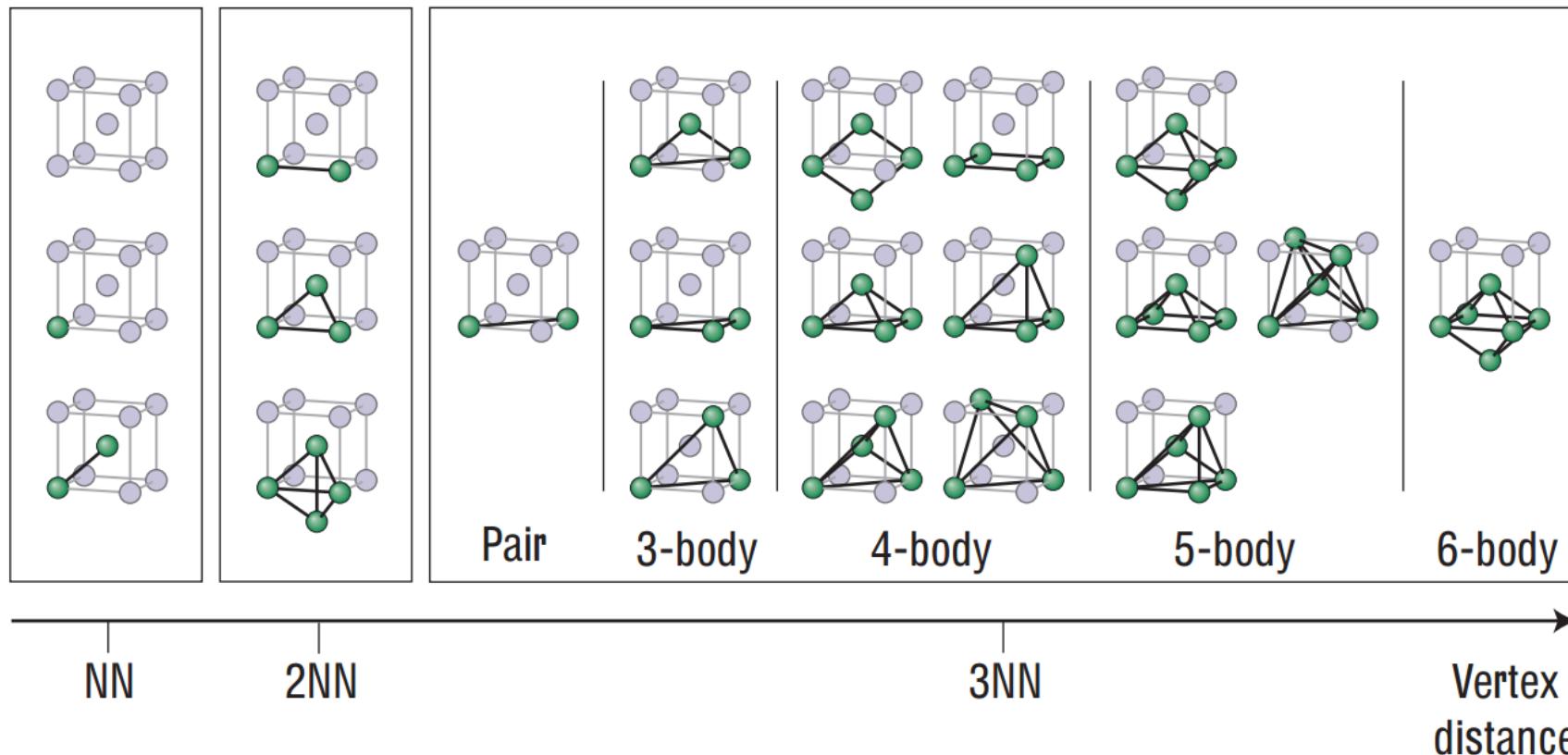
Department of Physics and Astronomy, Brigham Young University, Provo, Utah 84602, USA

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$$E(\sigma) = E_0 + \sum_f \bar{\Pi}_f(\sigma) J_f$$

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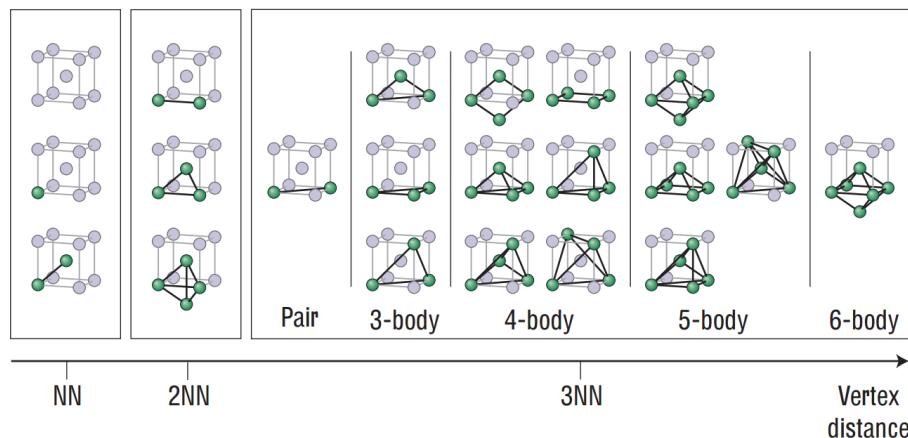
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$$E(\sigma) = E_0 + \sum_f \bar{\Pi}_f(\sigma) J_f$$



$$\min_u \mu \|\vec{u}\|_1 + \frac{1}{2} \|\mathbb{A}\vec{u} - \vec{f}\|^2$$

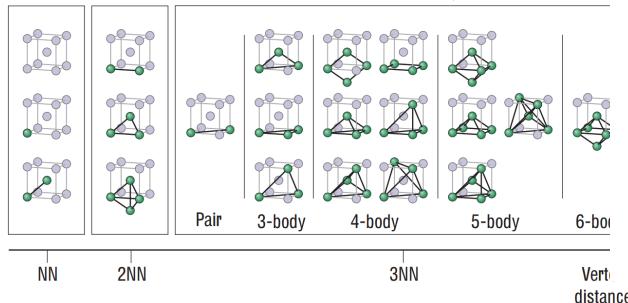
Bregman Iteration

$$\vec{f}^{k+1} = \vec{f} + (\vec{f}^k - \mathbb{A}\vec{u}^k),$$

$$\vec{u}^{k+1} = \arg \min_u \mu \|\vec{u}\|_1 + \frac{1}{2} \|\mathbb{A}\vec{u} - \vec{f}^{k+1}\|^2,$$

$$E(\sigma) = E_0 + \sum_f \bar{\Pi}_f(\sigma) J_f$$

PHYSICAL REVIEW B 87, 035125 (2013)



Compressive sensing as a paradigm for building physics models

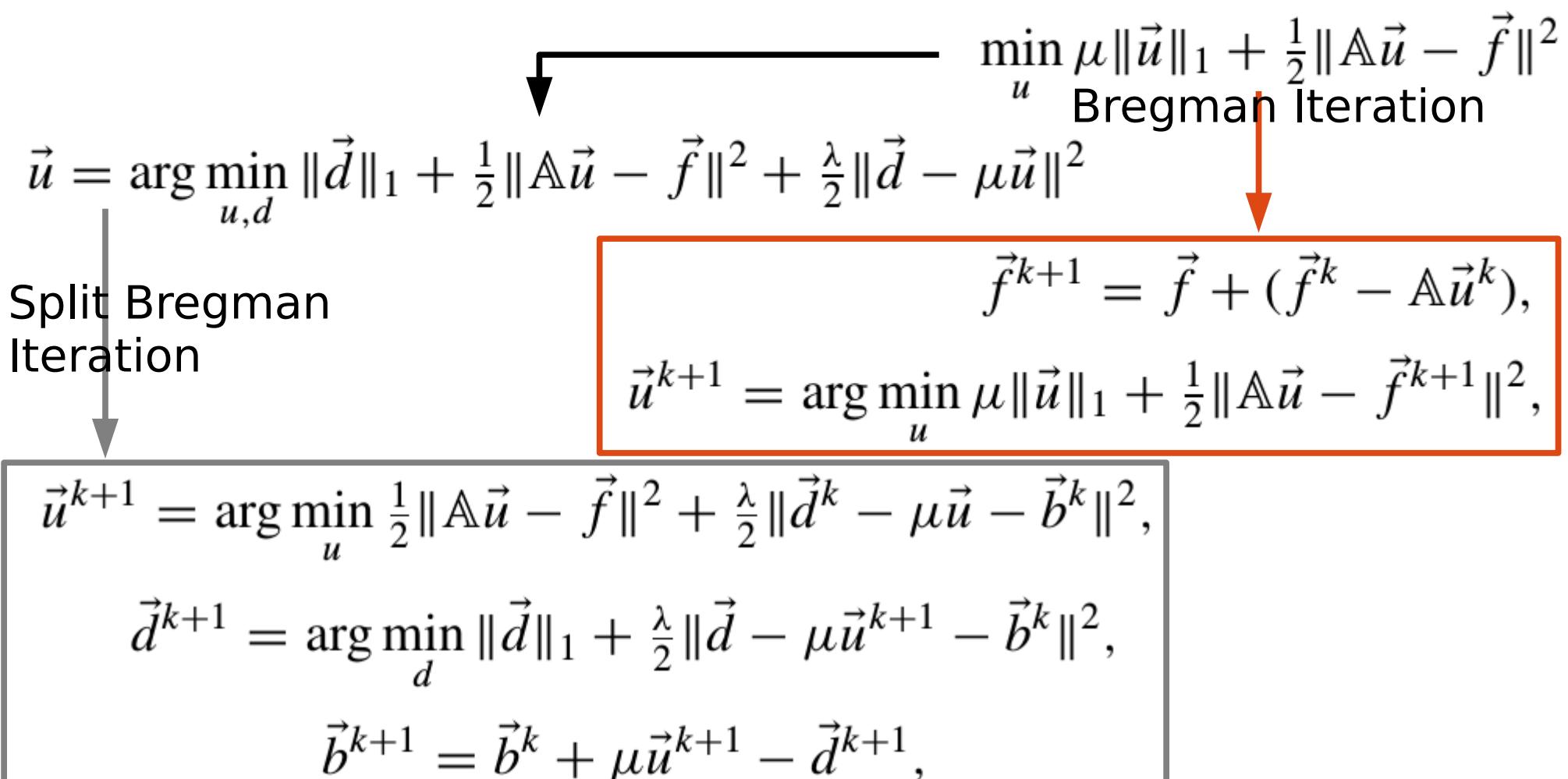
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Compressed modes for variational problems in mathematics and physics

Vidvuds Ozoliņš^{a,1}, Rongjie Lai^{b,1}, Russel Caflisch^{c,1}, and Stanley Osher^{c,1,2}

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Contributed by Stanley Osher, October 8, 2013 (sent for review September 3, 2013)

$$E_0 = \min_{\Phi_N} \sum_{j=1}^N \left\langle \phi_j, \hat{H} \phi_j \right\rangle \quad \text{s.t.} \quad \left\langle \phi_j, \phi_k \right\rangle = \delta_{jk}.$$

$$W_j(\mathbf{x}) = \sum_k U_{jk} \phi_k(\mathbf{x})$$

$$\left\langle \Delta \mathbf{x}_j^2 \right\rangle = \left\langle W_j, (\mathbf{x} - \langle \mathbf{x}_j \rangle)^2 W_j \right\rangle \quad \langle \mathbf{x}_j \rangle = \langle W_j, \mathbf{x} W_j \rangle$$

Parameter free maximally localised Wannier functions?

$$E = \min_{\Psi_N} \sum_{j=1}^N \left(\frac{1}{\mu} |\psi_j|_1 + \left\langle \psi_j, \hat{H} \psi_j \right\rangle \right) \quad \text{s.t.} \quad \left\langle \psi_j, \psi_k \right\rangle = \delta_{jk}$$

Lattice Anharmonicity and Thermal Conductivity from Compressive Sensing of First-Principles Calculations

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(Received 22 April 2014; published 27 October 2014)

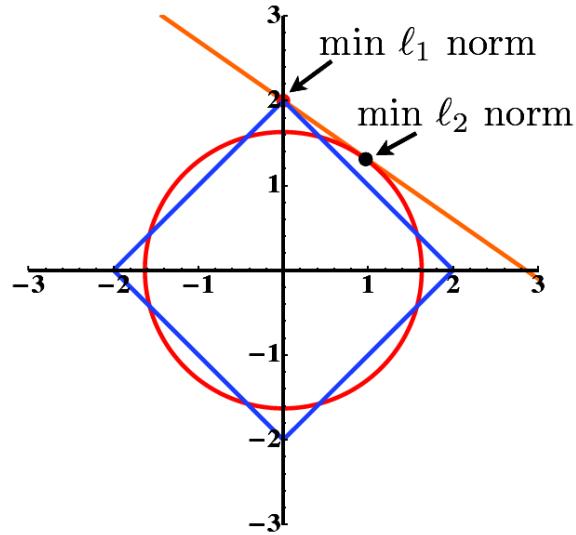
$$V = V_0 + \Phi_{\mathbf{a}} u_{\mathbf{a}} + \frac{\Phi_{\mathbf{ab}}}{2} u_{\mathbf{a}} u_{\mathbf{b}} + \frac{\Phi_{\mathbf{abc}}}{3!} u_{\mathbf{a}} u_{\mathbf{b}} u_{\mathbf{c}} + \dots,$$

$$\Phi_{\mathbf{ab}} \equiv \Phi_{ij}(ab) = \partial^2 V / \partial u_{\mathbf{a}} \partial u_{\mathbf{b}}$$

$$\Phi_{\mathbf{abc}} \equiv \Phi_{ijk}(abc) = \partial^3 V / \partial u_{\mathbf{a}} \partial u_{\mathbf{b}} \partial u_{\mathbf{c}}$$

$$\begin{aligned} \Phi^{\text{CS}} &= \arg \min_{\Phi} \|\Phi\|_1 + \frac{\mu}{2} \|\mathbf{F} - \mathbb{A}\Phi\|_2^2 \\ &= \arg \min_{\Phi} \sum_I |\Phi_I| + \frac{\mu}{2} \sum_{ai} (F_{ai} - A_{ai,J} \Phi_J)^2 \end{aligned}$$

Compressed sensing: the quest for descriptors and predictive models

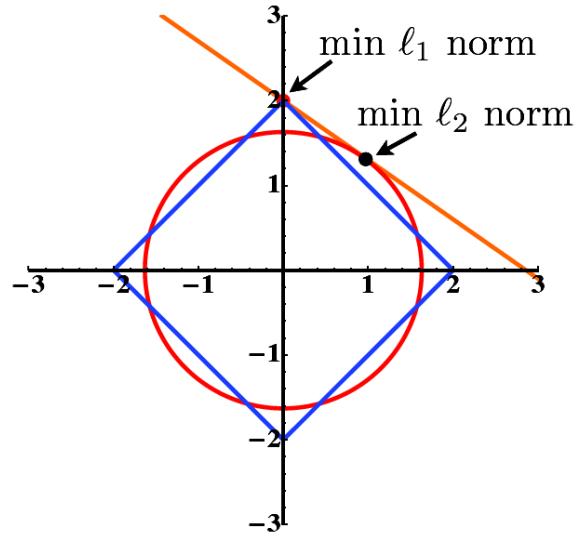


$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_1$$

(Possibly) optimal solution
Convex optimization
Moderate # columns in \mathbf{D}

$\|\mathbf{c}\|_1$ “Manhattan”. Sum of absolute values of the elements of \mathbf{c}

Compressed sensing: the quest for descriptors and predictive models



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When there are high correlations,
LASSO+ ℓ_0 (LMG et al. PRL 2015):

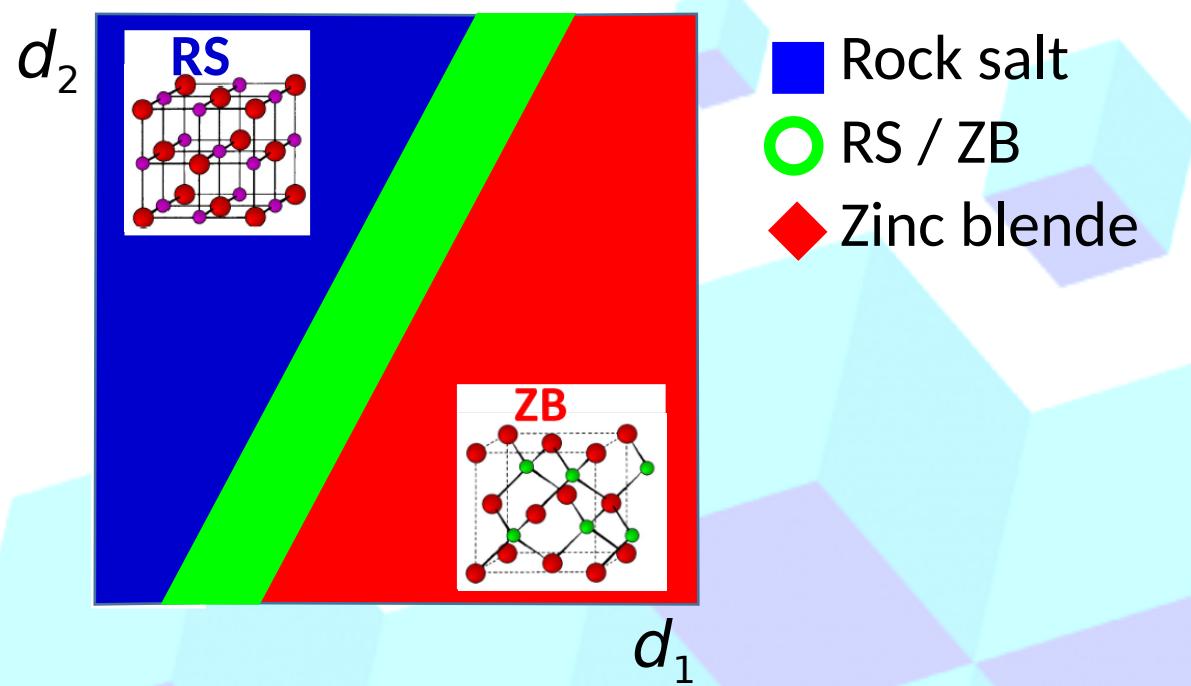
- use LASSO with lambda in order to “switch on” few tens features (say 30-50)
- perform ℓ_0 regularization, i.e., for 1,2,3D solution, enumerate all 1- 2- 3-tuples and find the best fitting tuple.

Compressed sensing: the quest for descriptors and predictive models

82 octet AB binary compounds

Ansatz: atomic features

- HOMO
- LUMO
- Ionization Potential
- Electron Affinity
- Radius of valence *s* orbital
- Radius of valence *p* orbital
- Radius of valence *d* orbital
- Billions of non-linear functions of the above



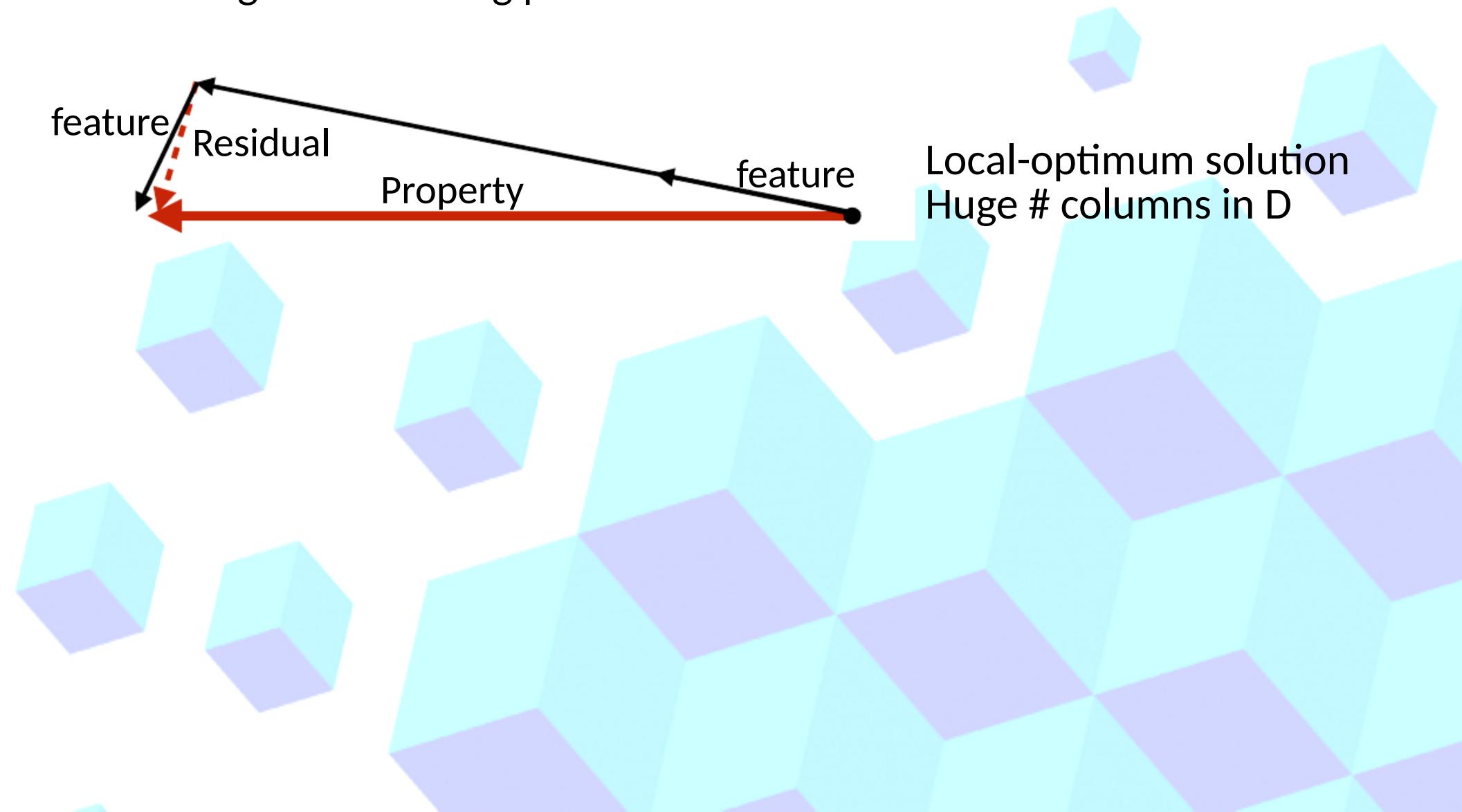
$$\mathbf{P} = c_1 \mathbf{d}_1 + c_2 \mathbf{d}_2 + \dots + c_n \mathbf{d}_n$$

$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0$$

$E(\text{Rock salt}) - E(\text{Zinc blende})$

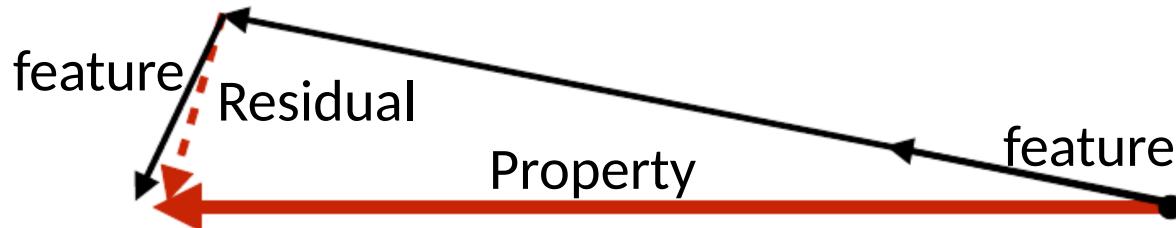
Compressed sensing: the quest for descriptors and predictive models

From orthogonal matching pursuit



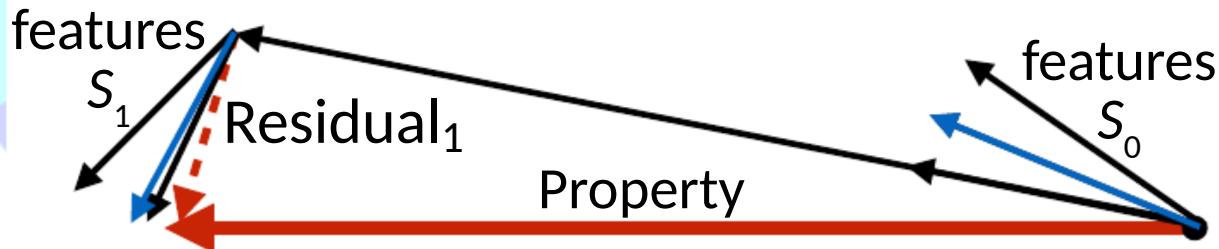
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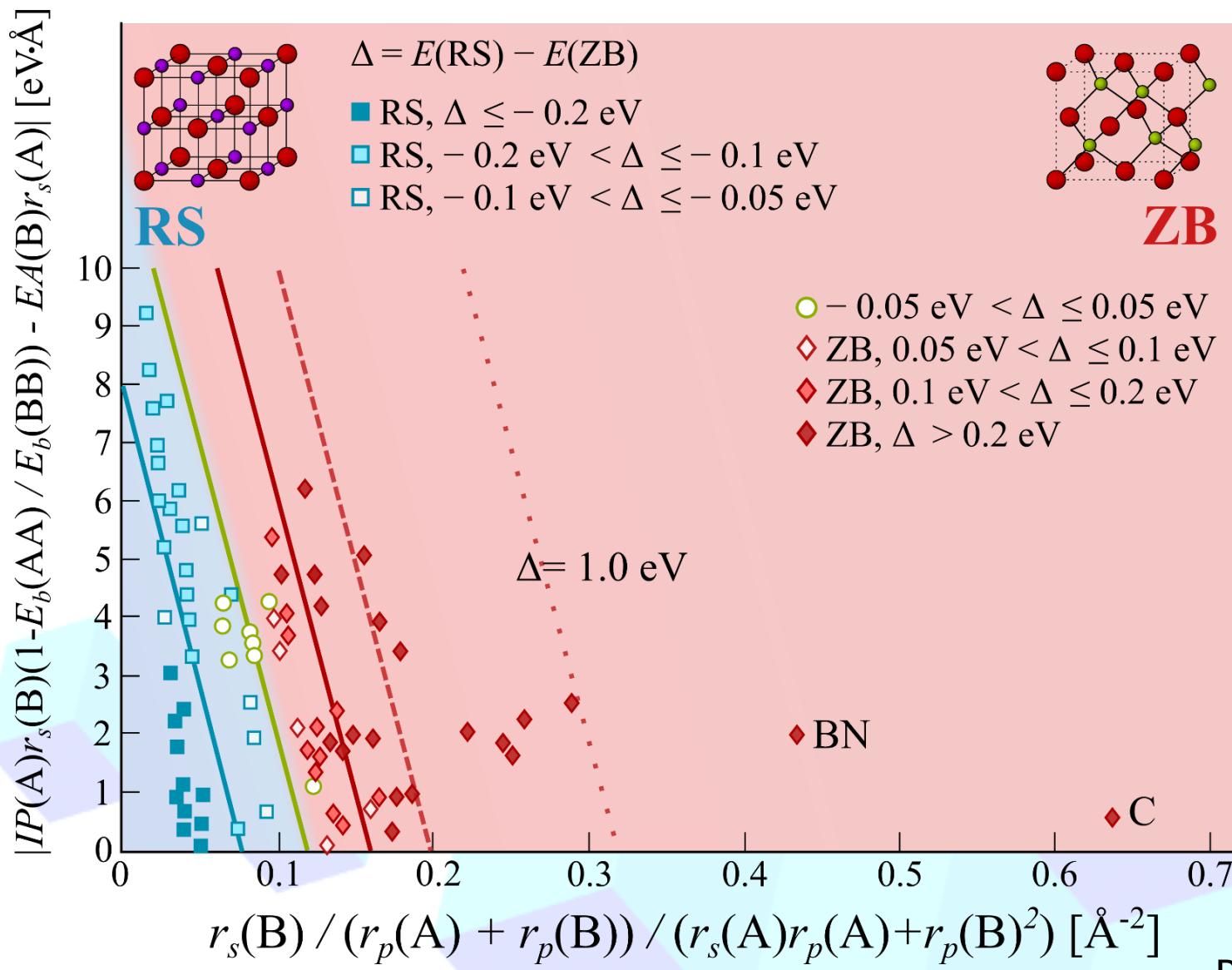
Local-optimum solution
Huge # columns in D

... to Sure Independence Screening + Sparsifying Operator (SISSO)



Proxy of
global-optimum solution
Huge # columns in D

Compressed sensing: the quest for descriptors and predictive models



One descriptor to rule them all: Multi-task learning

$$\{P^{(1)}, P^{(2)}, \dots, P^{N^T}\}$$



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$$\{P^{(1)}, P^{(2)}, \dots, P^{N^T}\} \quad \longrightarrow \quad P^k = \mathbf{d} \cdot \mathbf{c}^k$$



One descriptor to rule them all: Multi-task learning

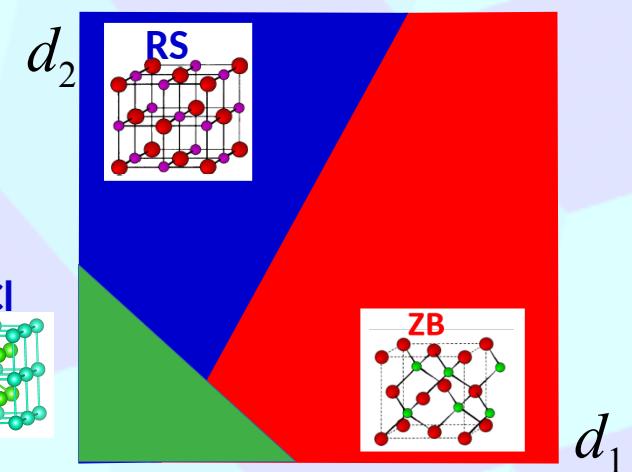
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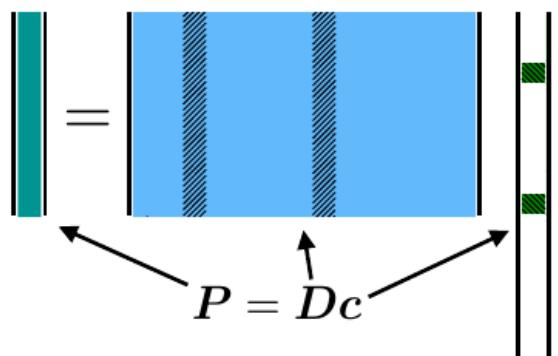
Application: multi-phase stability diagram
Properties: crystal-structure formation energies

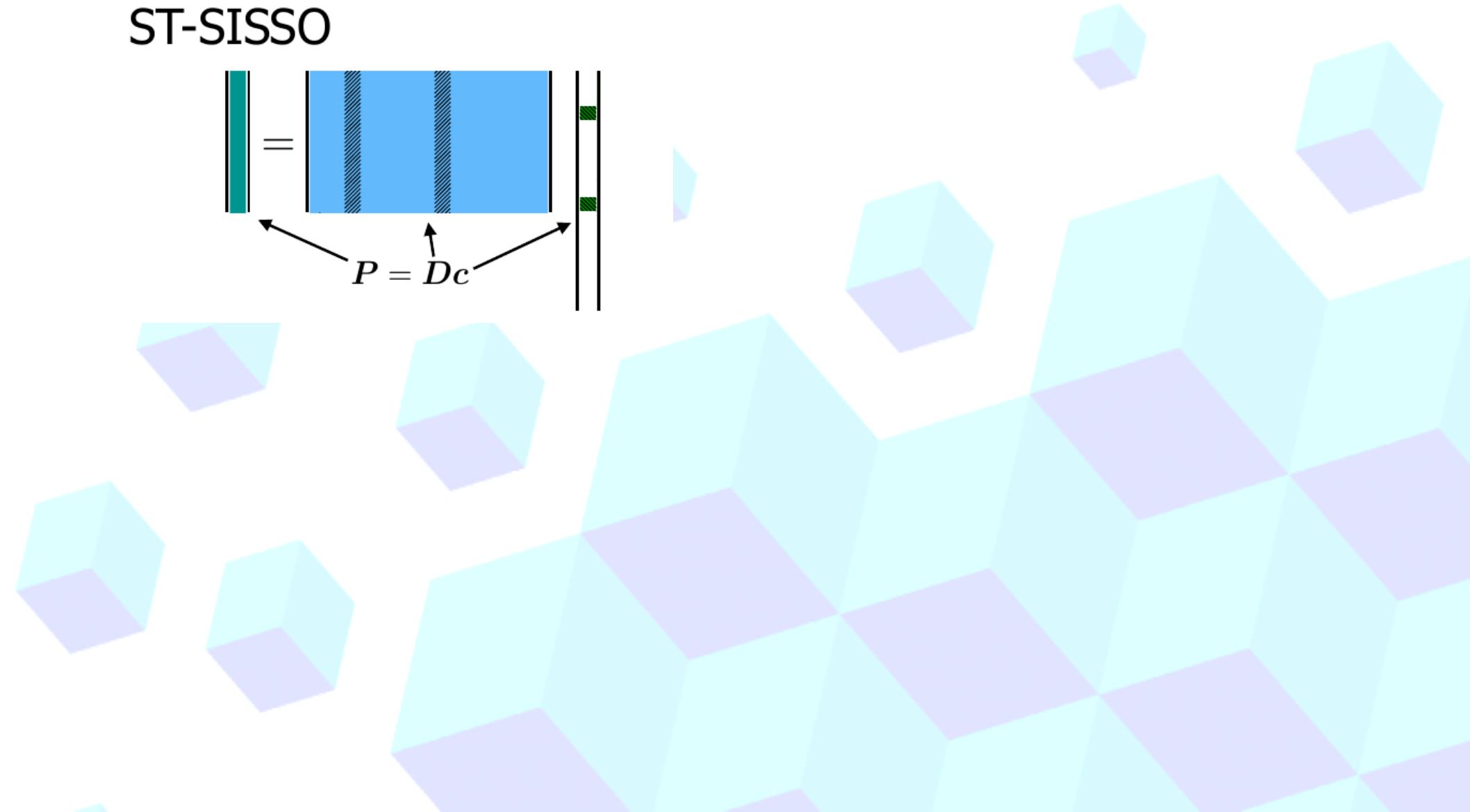


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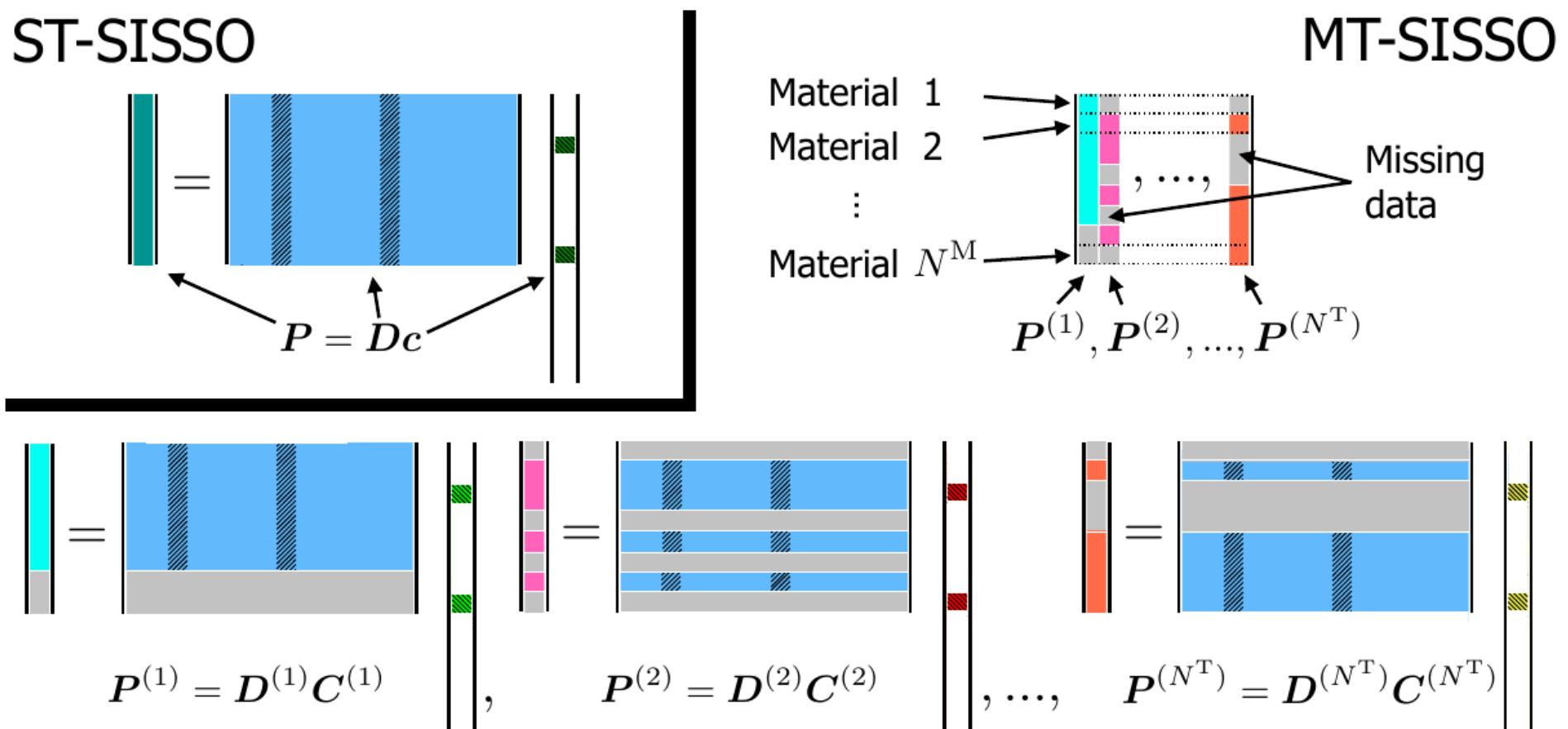
ST-SISSO

$$\mathbf{P} = \mathbf{D} \mathbf{c}$$




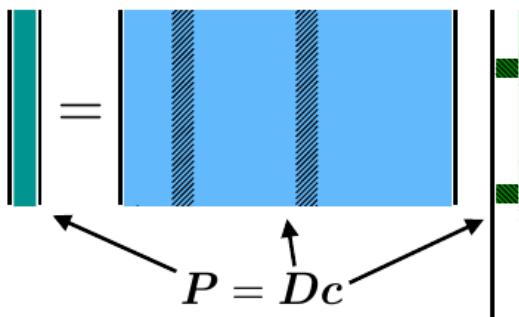
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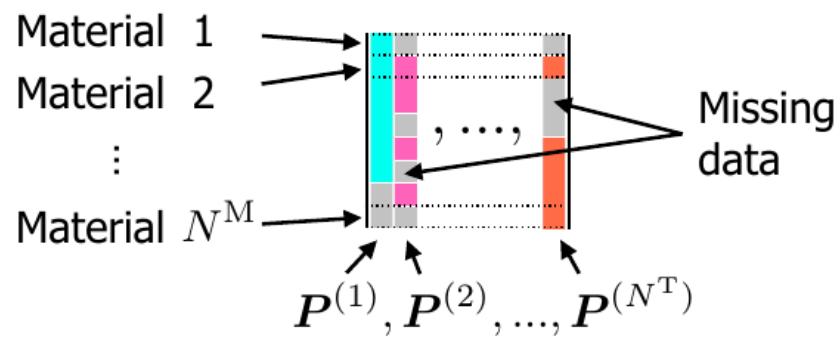


One descriptor to rule them all: Multi-task learning

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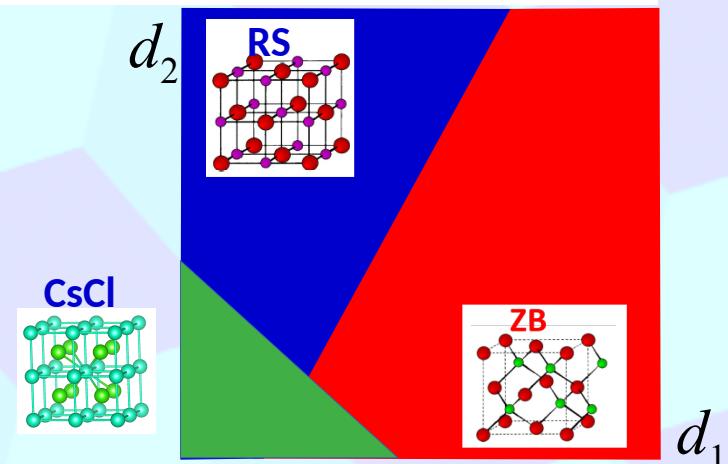


MT-SISSO



$P^{(1)} = D^{(1)}C^{(1)}$, $P^{(2)} = D^{(2)}C^{(2)}$, ..., $P^{(N^T)} = D^{(N^T)}C^{(N^T)}$

Application: multi-phase stability diagram
Properties: crystal-structure formation energies



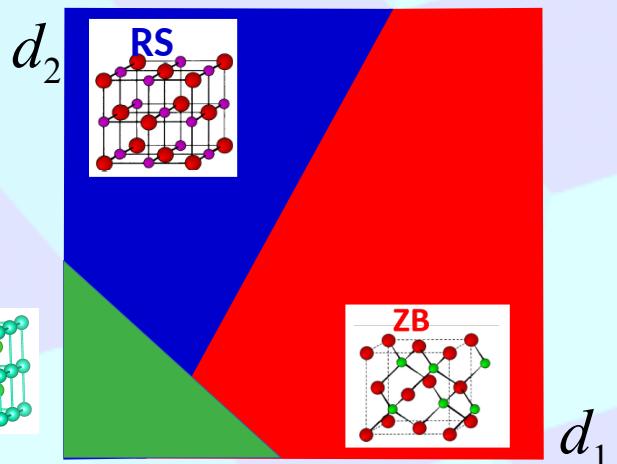
One descriptor to rule them all: Multi-task learning

$$\{P^{(1)}, P^{(2)}, \dots, P^{N^T}\} \rightarrow P^k = \mathbf{d} \cdot \mathbf{c}^k$$

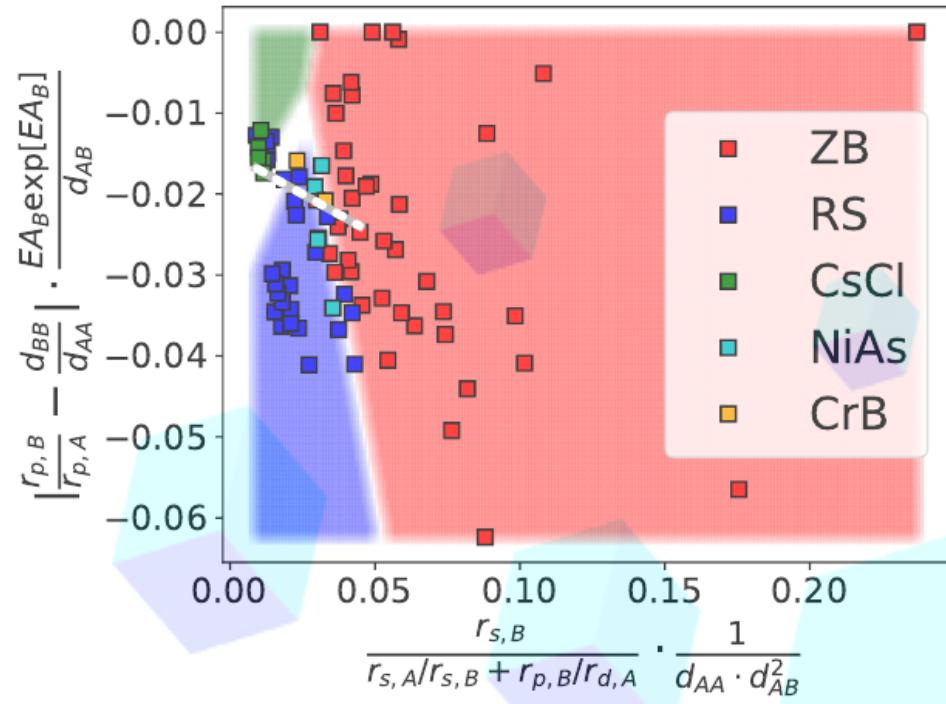
$$\arg \min_{\mathbf{c}} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

$$\arg \min_{\mathbf{C}} \sum_{k=1}^{N^T} \frac{1}{N_k^M} \|\mathbf{P}^k - \mathbf{D}^k \mathbf{C}^k\|_2^2 + \lambda \|\mathbf{C}\|_0$$

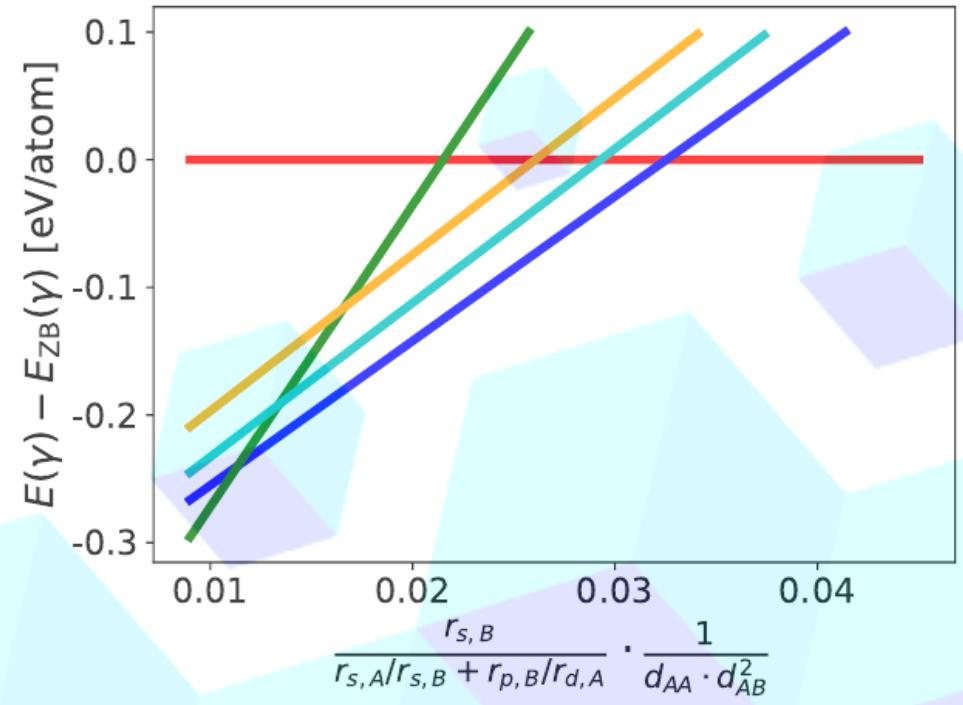
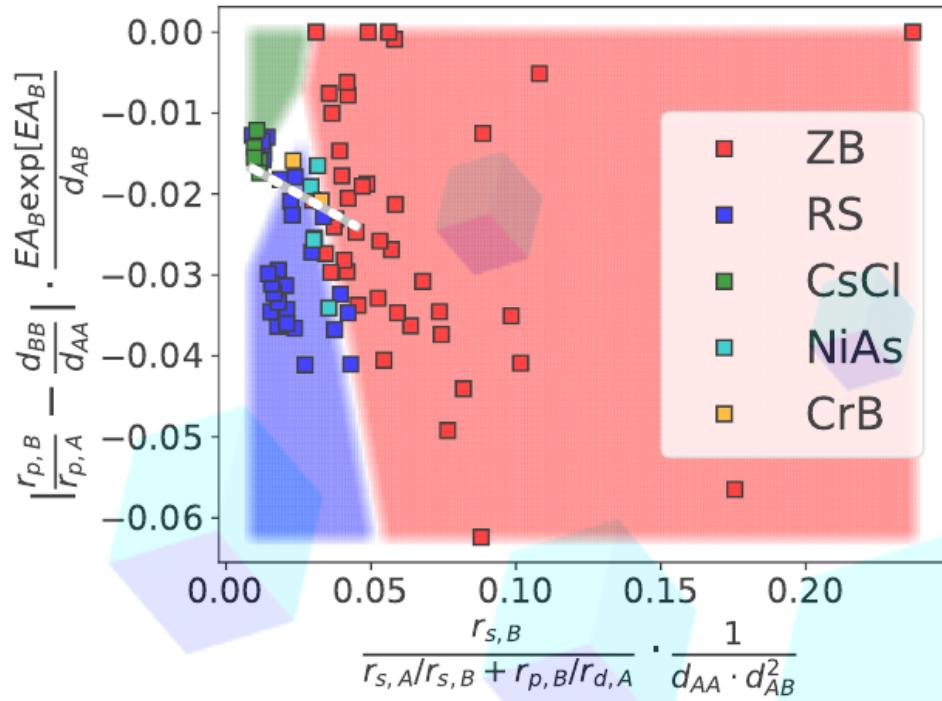
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One descriptor to rule them all: Multi-task SISSO Energy differences among 5 crystal structures.



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A general scheme about training, cross-validation, test

Training: input (features, descriptor) + labels (values target property)
→ yields one model which minimizes a cost function (incl.
regularization)



A general scheme about training, cross-validation, test

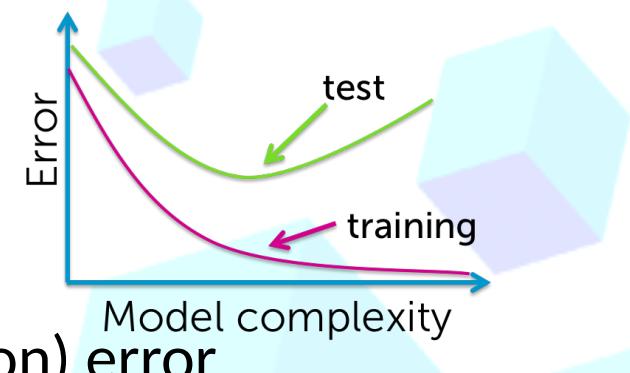
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Cross-validation: used to tune model-complexity

- perform training n times on different split of data.

Training + test/validation sets

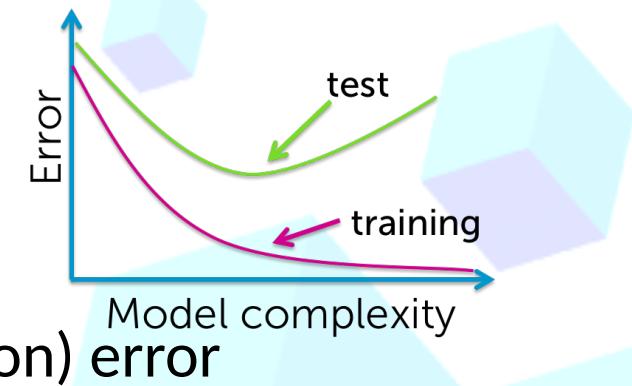
→ yields one model that minimizes the test (validation) error



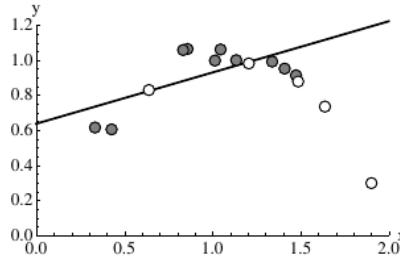
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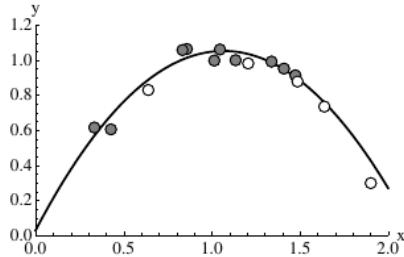
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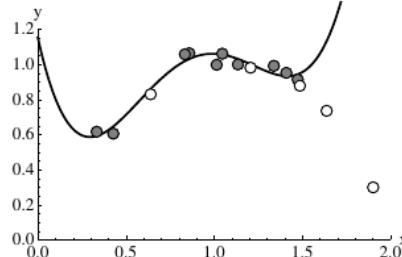
Underfitting



Fitting



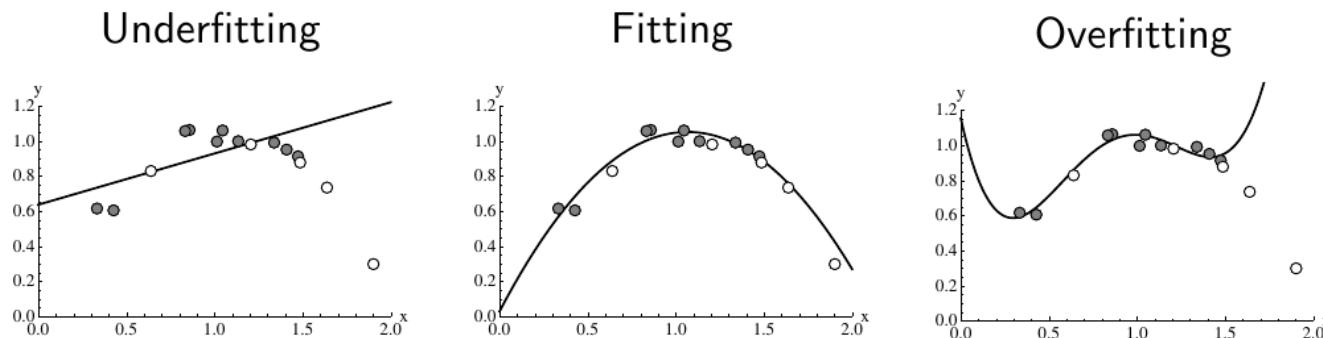
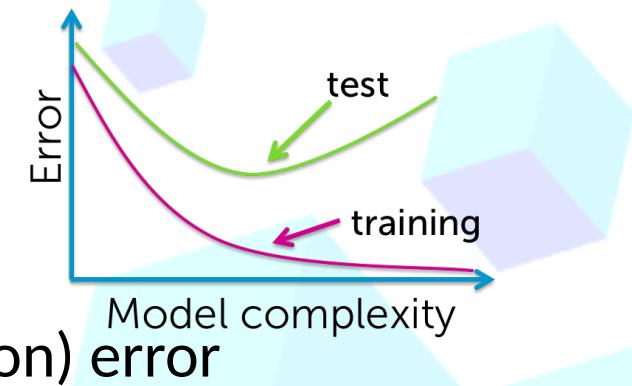
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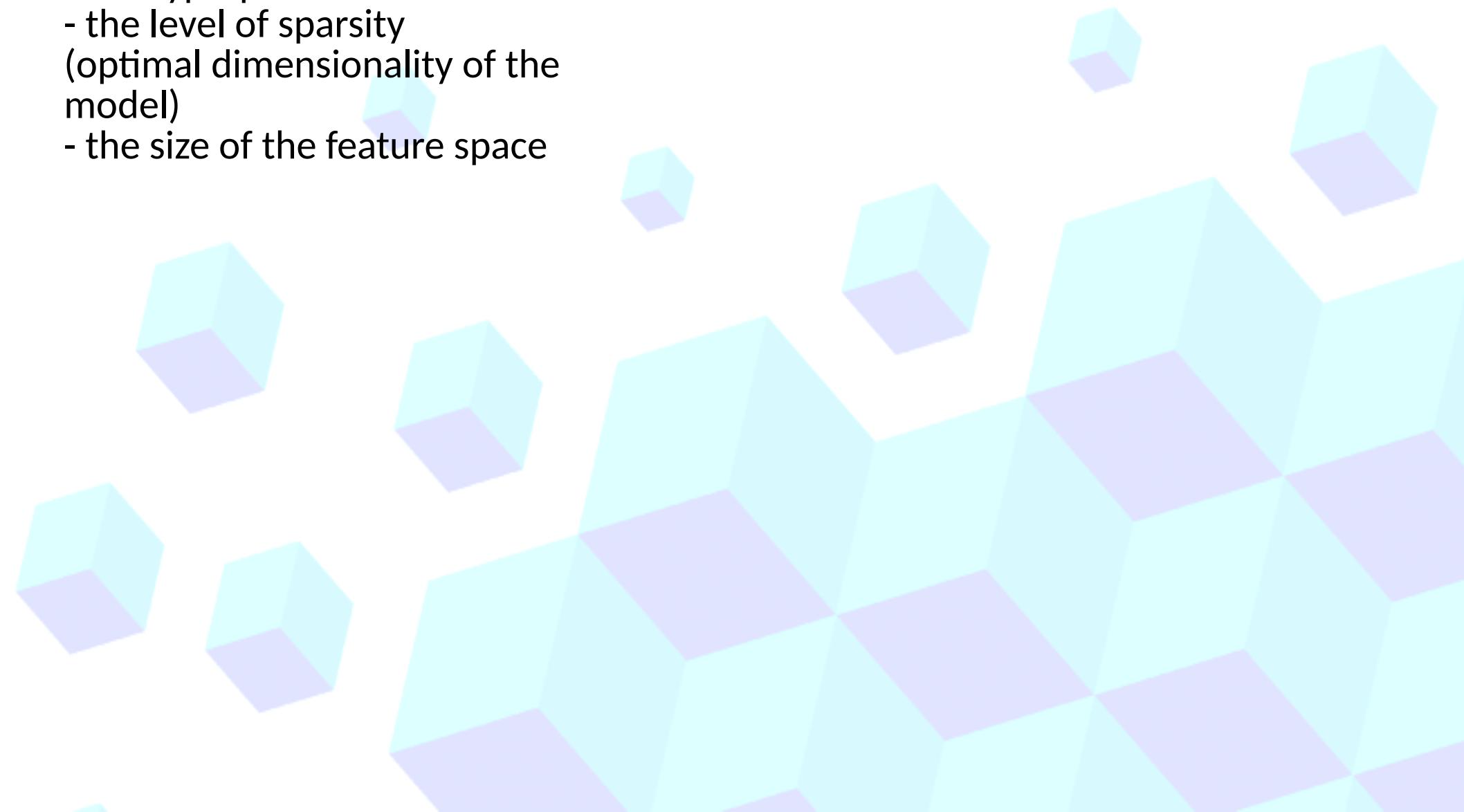
Cross-validation: used to tune model-complexity
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Training + test/validation sets
→ yields one model that minimizes the test (validation) error



Test: evaluation of the performance of the model on data never used for training (i.e., the whole cross-validation procedure), aka left-out set

Data-driven model complexity

- In compressed sensing the “hyperparameters” are
 - the level of sparsity (optimal dimensionality of the model)
 - the size of the feature space

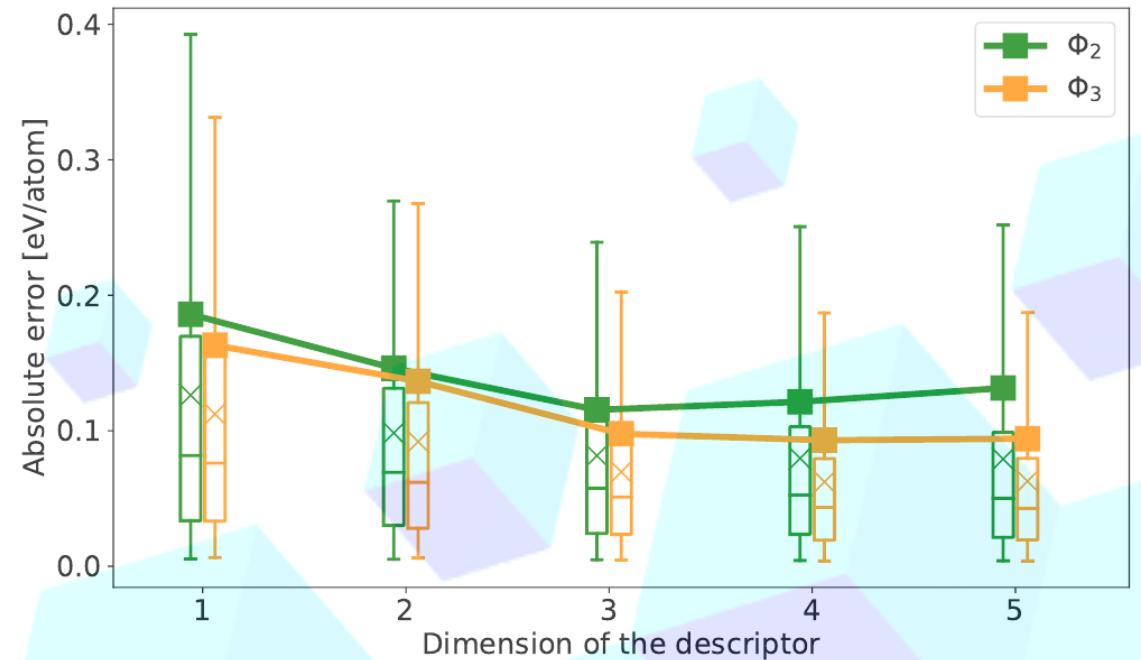


Data-driven model complexity

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- Tuned via cross-validation:
Iterated random selection of a subset of the data for training and test on the left out set

Data-driven model complexity

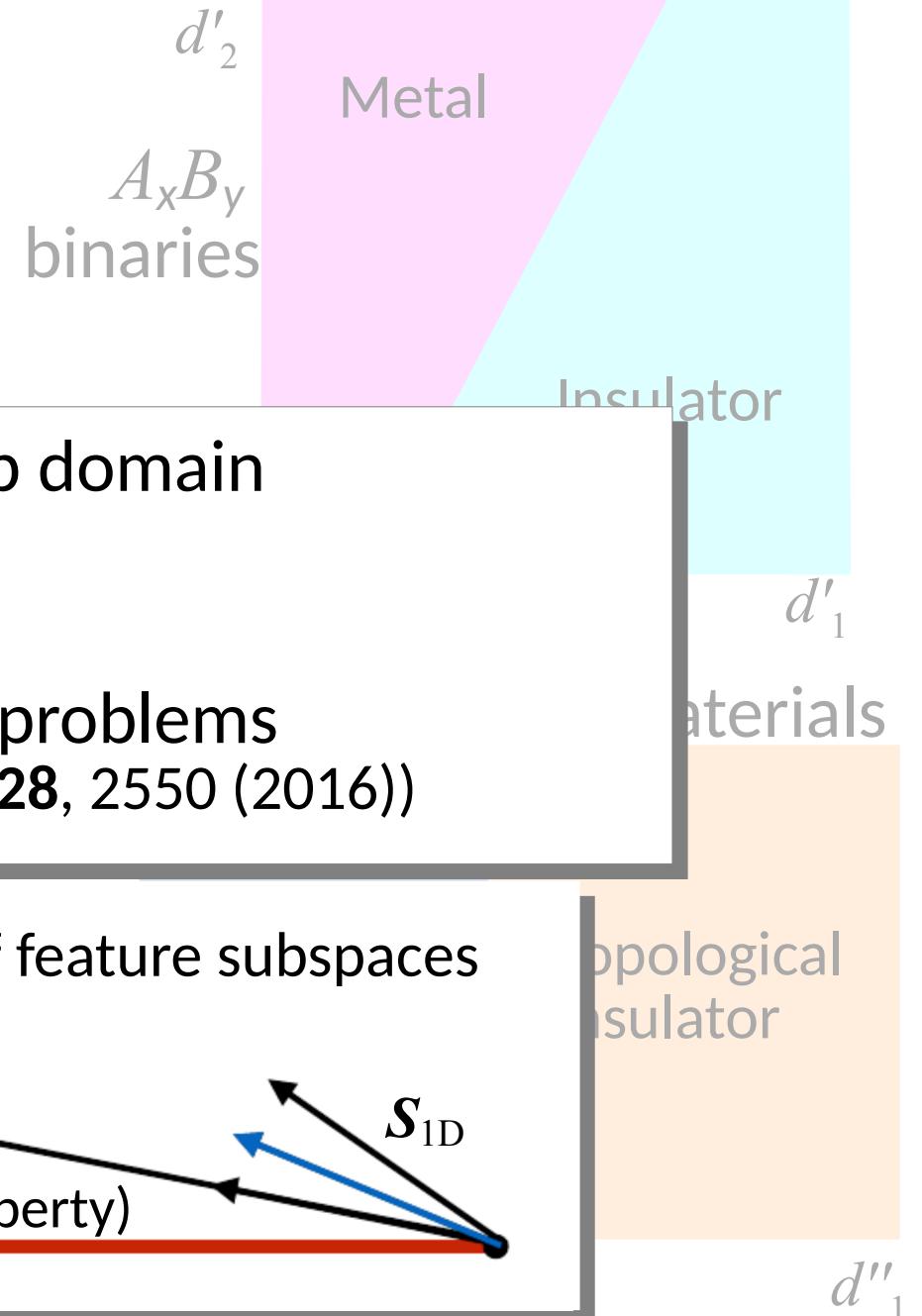
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Charts/maps of materials

$$\underset{\mathbf{c} \in \mathbb{R}^M}{\operatorname{argmin}} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

New cost function to be minimized:
overlap of convex domains



SISSO: metal/nonmetal classification of binary materials

Challenge:

Given the formula A_xB_y of a binary material AND its crystal structure, **is it a metal or a nonmetal?**

Dataset:

~300 materials from *Springer Materials*

B is a *p*-block element, A any element

3D materials (i.e., not layered)

At least one 1st neighbor of $A(B)$ is $B(A)$

(i.e., no materials containing “clusters” of A and/or B)

Classification AND primary features from *experiments*:

ionization energy, electron affinity,

(Pauling) electronegativity,

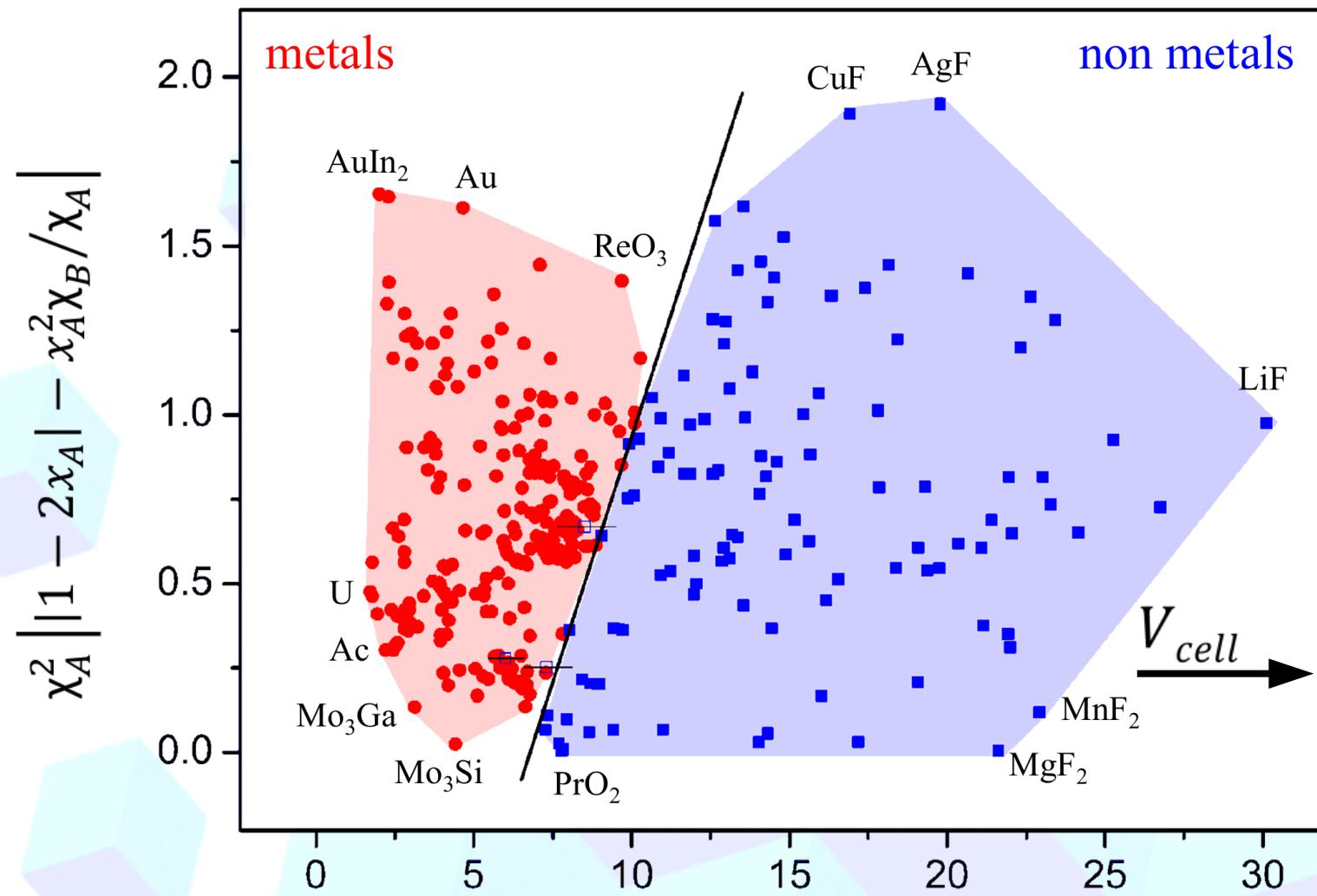
covalent radius,

valence, atomic fraction,

AB interatomic distance,

cell volume normalized by the sum of atomic volumes

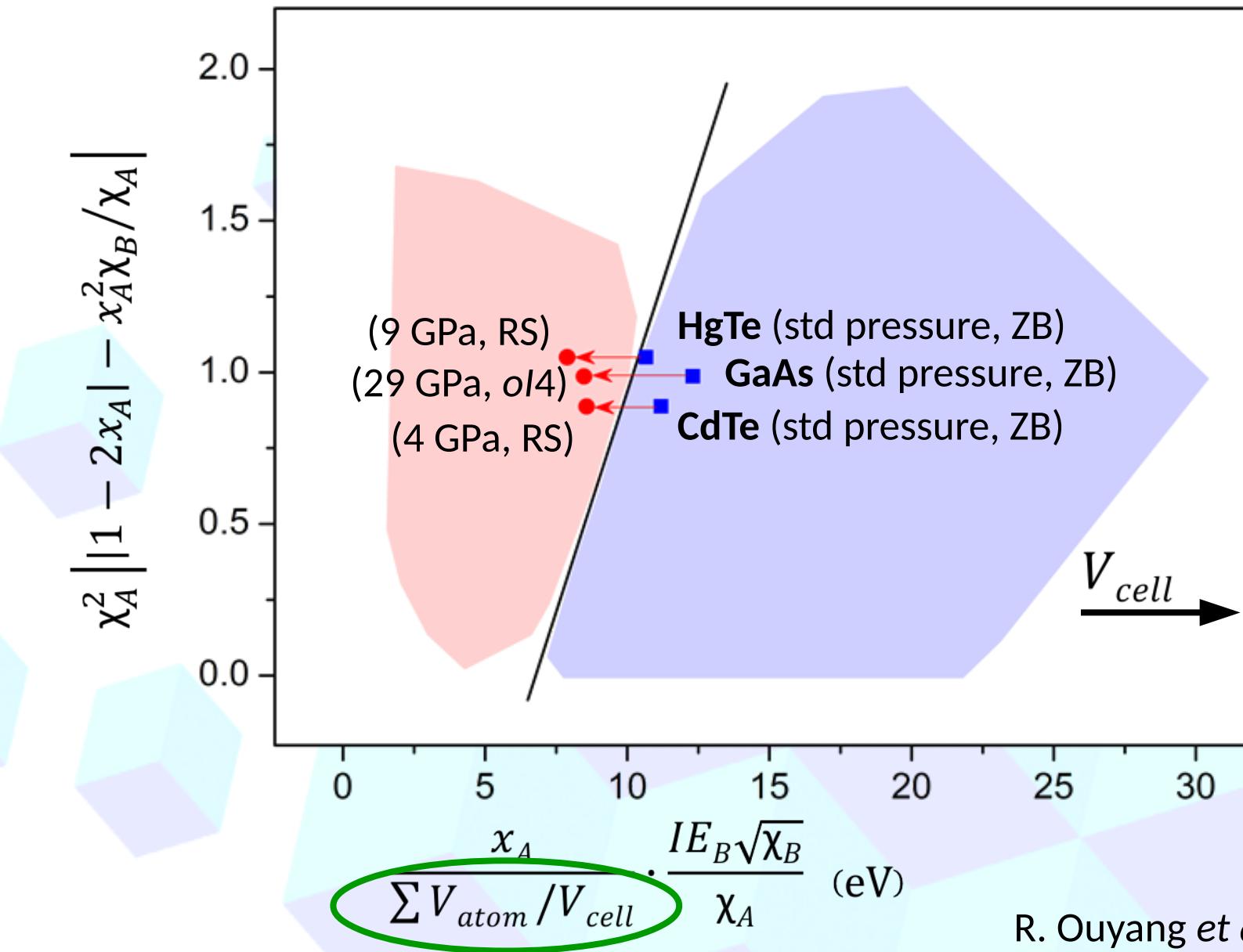
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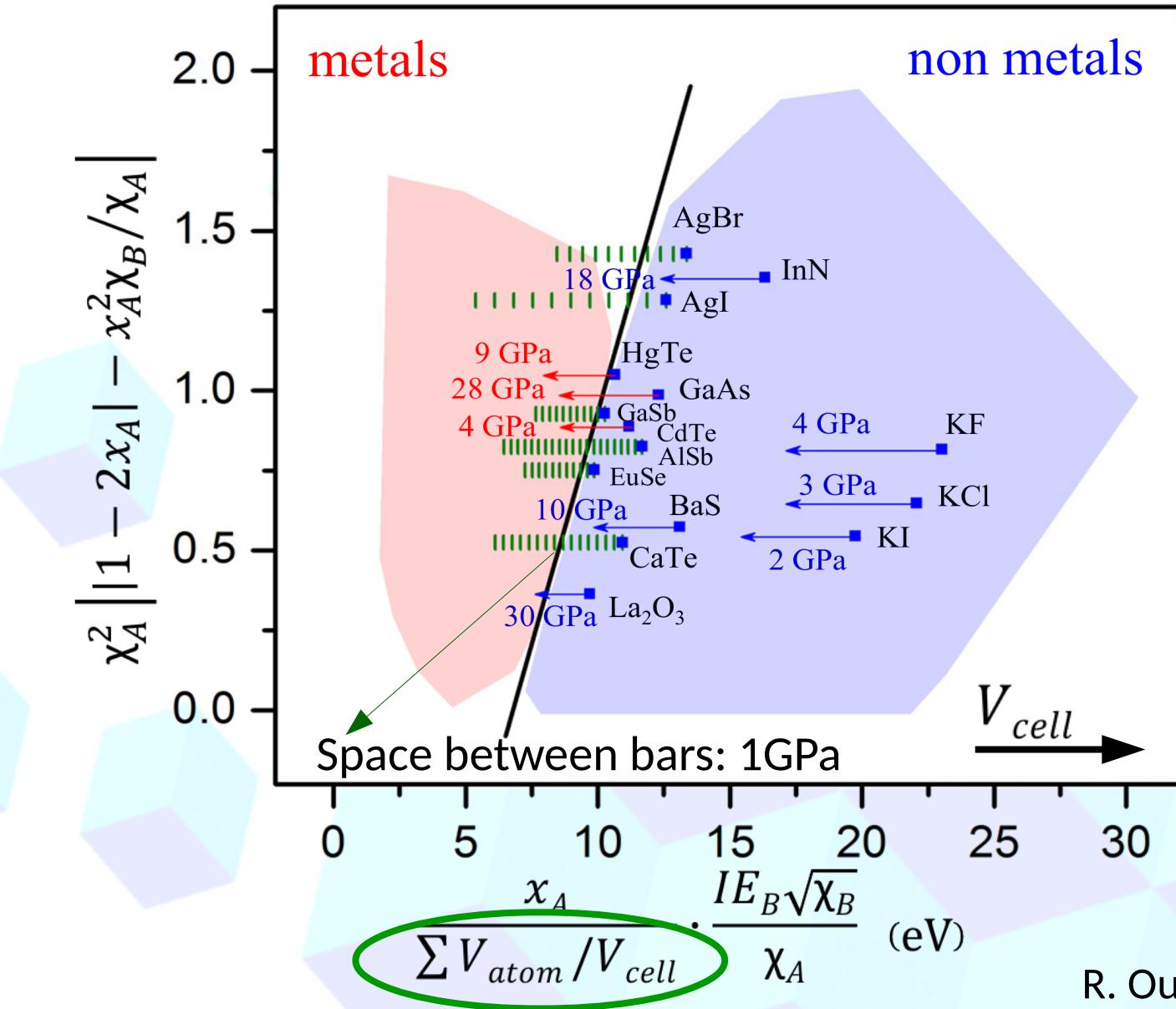
x Atomic fraction
 IE Ionization energy
 χ Electronegativity

$$\frac{\sum V_{atom} / V_{cell}}{x_A} \cdot \frac{IE_B \sqrt{\chi_B}}{\chi_A} \text{ (eV)}$$

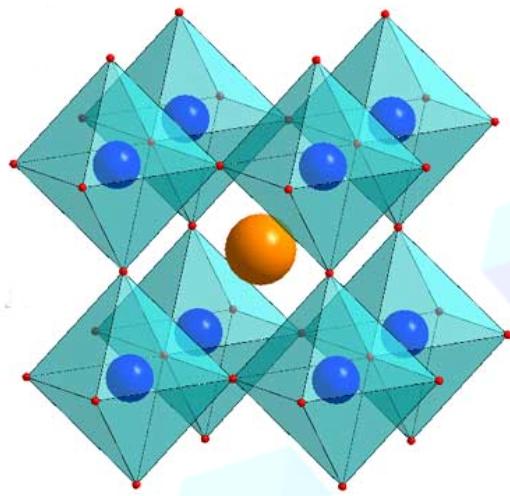
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Perovskites' stability: an improved Goldschmidt Tolerance Factor

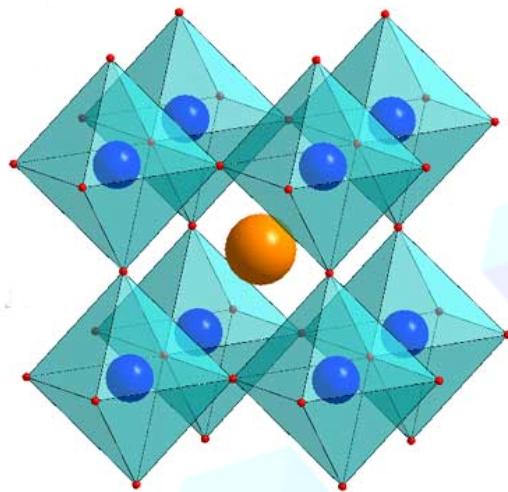


ABX_3

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \longrightarrow \text{Ionic radius}$$

Goldschmidt* stable perovskites: $0.825 < t < 1.059$, accuracy 79%

Perovskites' stability: an improved Goldschmidt Tolerance Factor



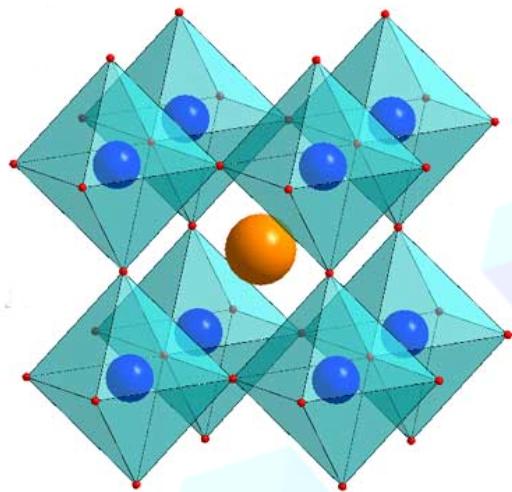
ABX_3

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$$\tau = \frac{r_X}{r_B} - n_A \left(n_A - \frac{r_A/r_B}{\ln(r_A/r_B)} \right) \rightarrow \begin{array}{l} \text{Oxidation state} \\ 1/\mu = \text{Octahedral factor} \end{array}$$

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Our stable perovskites: $\tau < 4.18$, accuracy 92%

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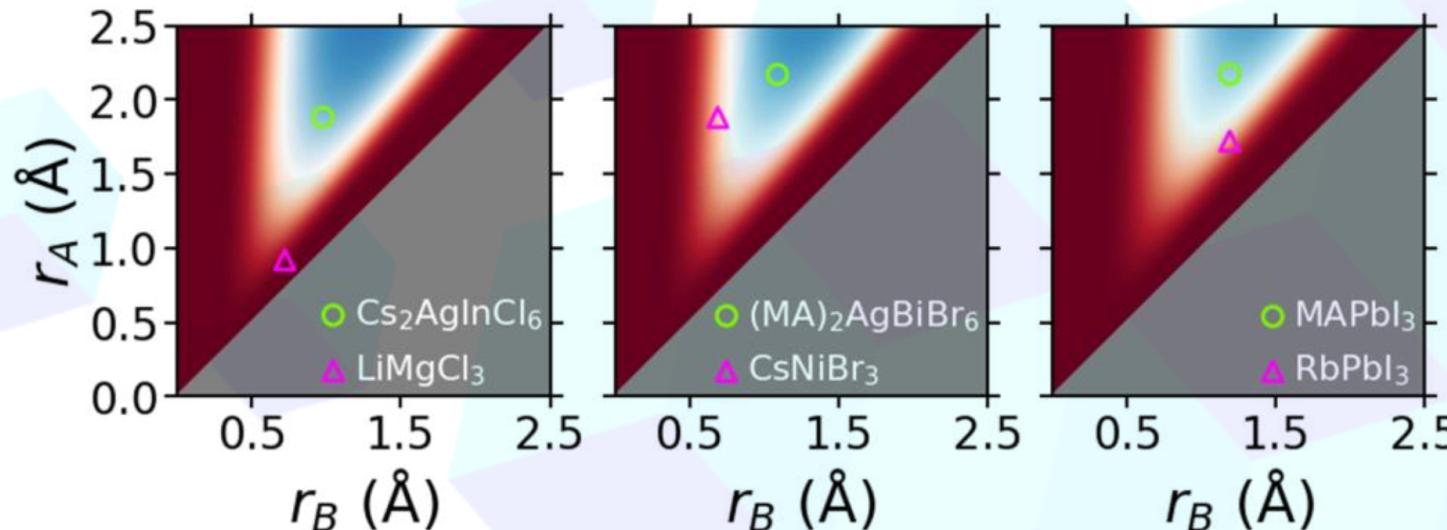
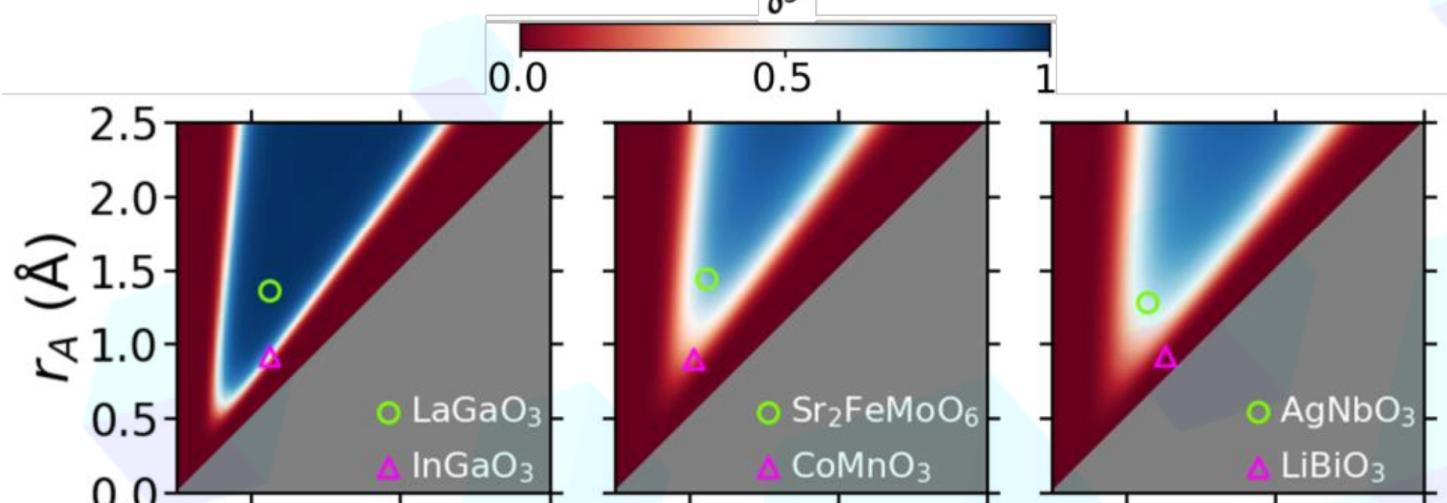
Our stable perovskites: $\tau < 4.18$, accuracy 92%

$\tau < 3.31$ or $\tau > 5.92$, 99% accuracy (1/3 of the training data)

$\tau < 3.31$ or $\tau > 12.08$, 100% accuracy (1/4 of the training data)

Improved Goldschmidt Tolerance Factor: Materials design

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \longrightarrow \tau = \frac{r_X}{r_B} - n_A \left(n_A - \frac{r_A/r_B}{\ln(r_A/r_B)} \right)$$

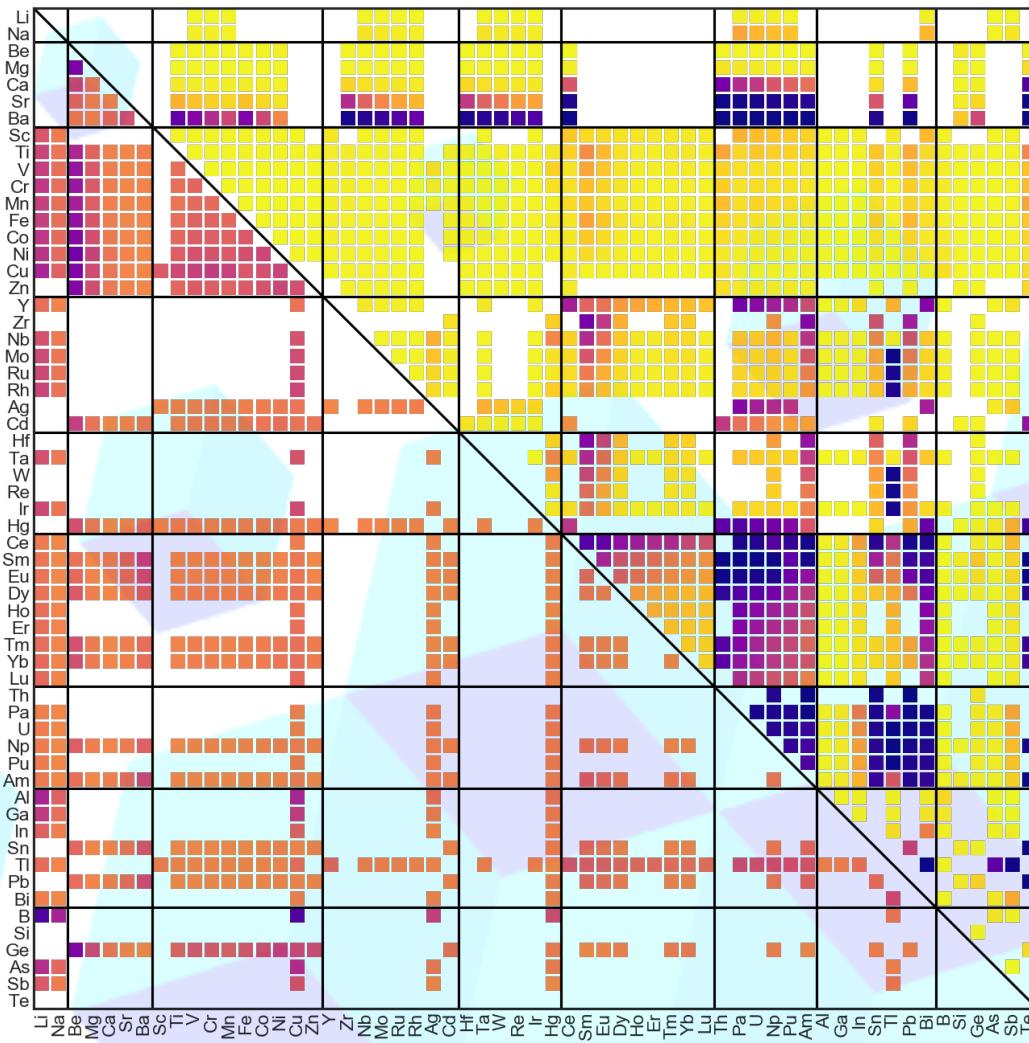
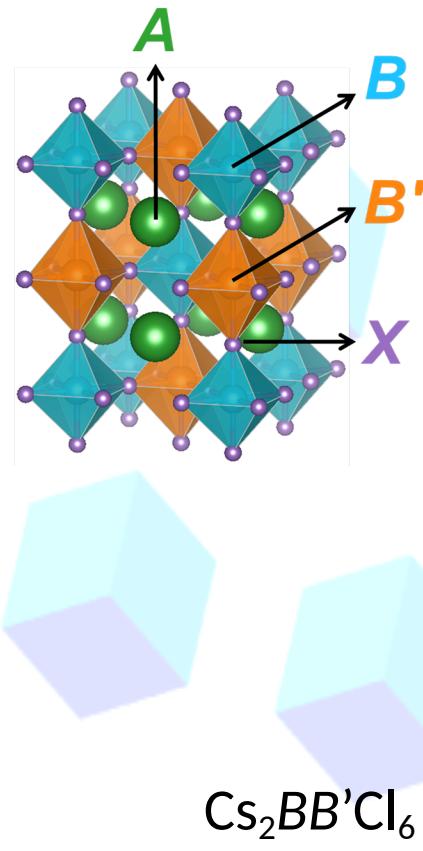


n_A $X = O^{2-}$

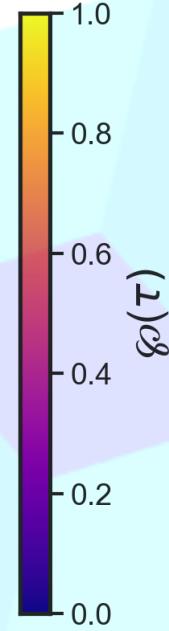
$n_A = 1^+$ X

Improved Goldschmidt Tolerance Factor: Extension of the materials space

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \quad \rightarrow \quad \tau = \frac{r_X}{r_B} - n_A \left(n_A - \frac{r_A/r_B}{\ln(r_A/r_B)} \right)$$

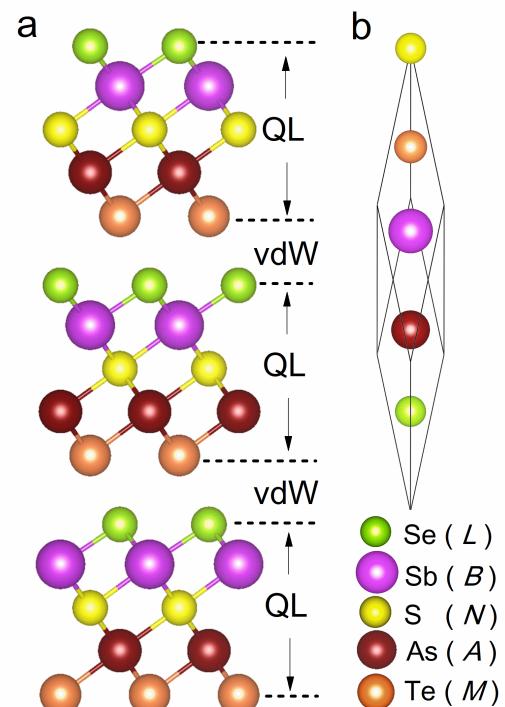


$\text{La}_2\text{BB}'\text{O}_6$



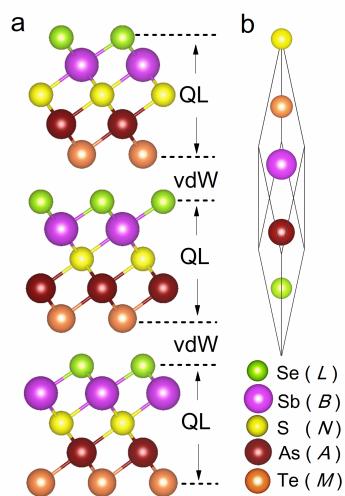
SISSO: predicting new tetradymite topological insulators

Prototype formula:
 $AB-LNM$
 $AB = \{As, Sb, Bi\}$
 $LNM = \{S, Se, Te\}$



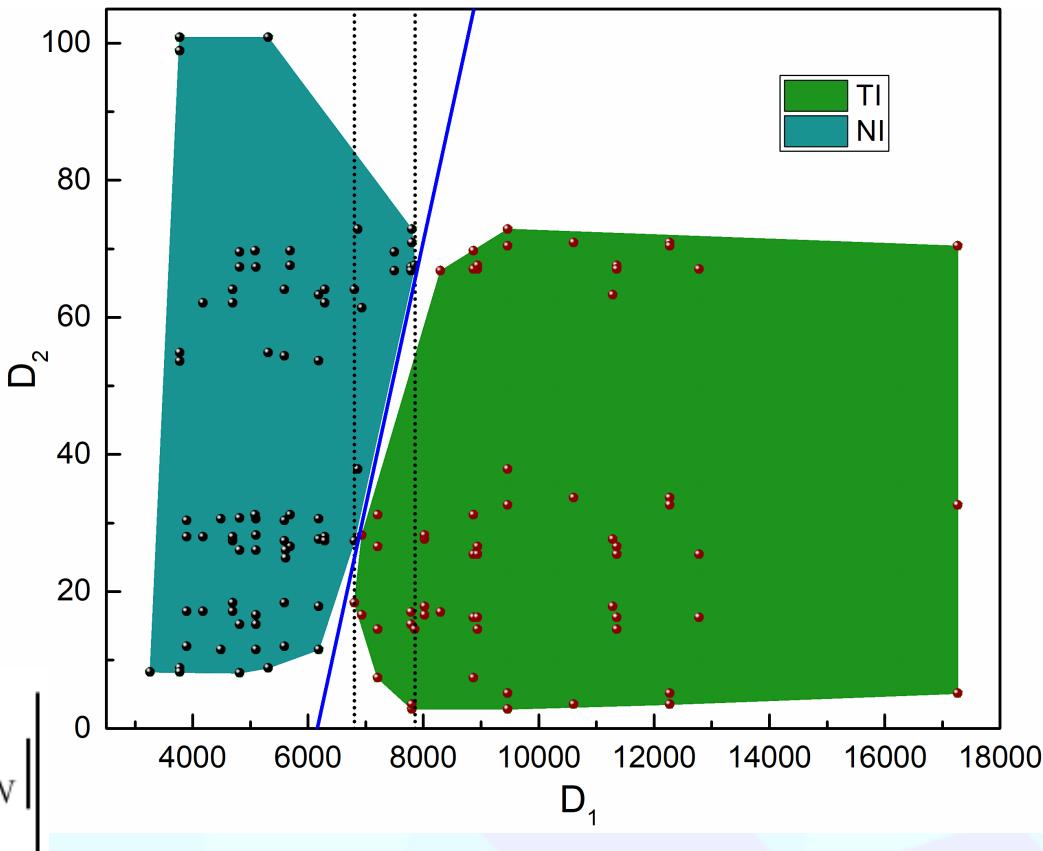
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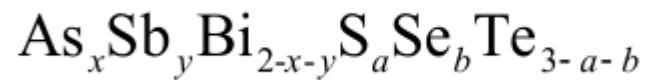
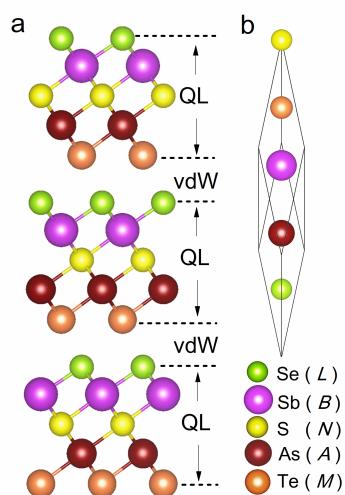
$$D_1 = (Z_A + Z_B) \cdot (Z_L + Z_M) - |Z_A Z_M - Z_B Z_L|$$

$$D_2 = \left| \frac{(\chi_M + \chi_N) \cdot Z_E}{\chi_A} - (Z_M + Z_N) - |Z_M - Z_N| \right|$$



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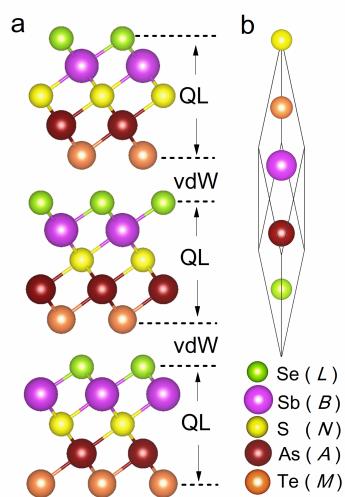
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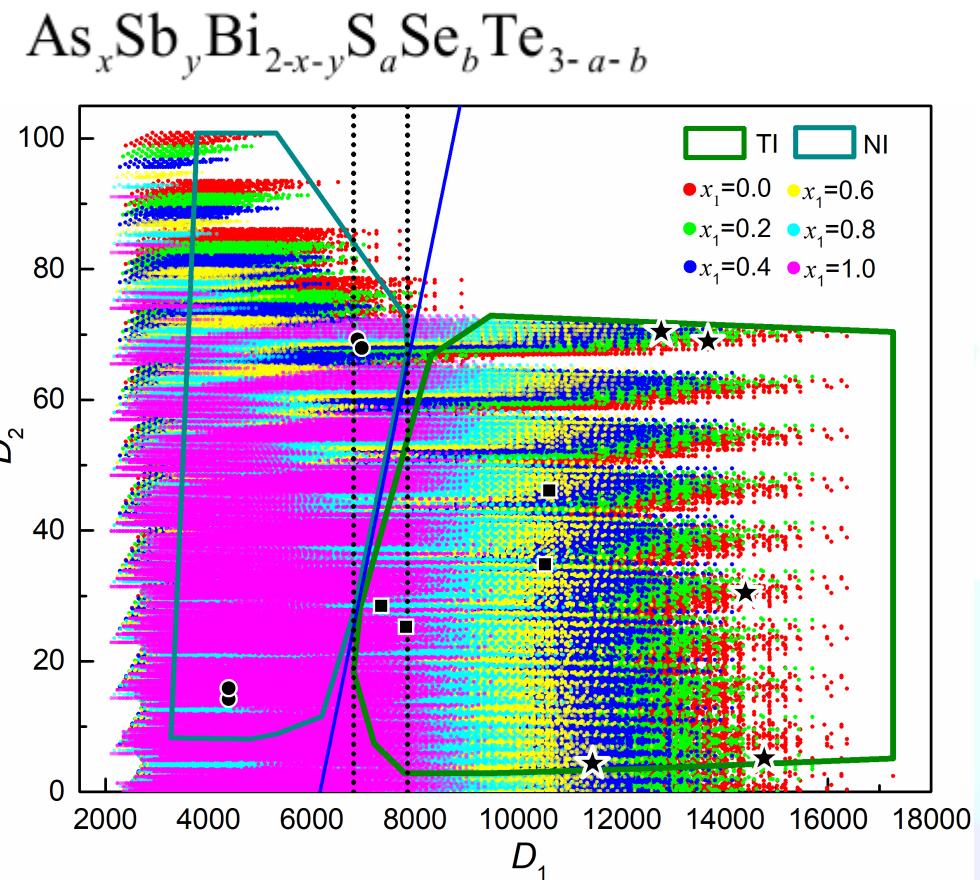
AB = {As, Sb, Bi}

LNM = {S, Se, Te}



$$D_1 = (Z_A + Z_B) \cdot (Z_L + Z_M) - |Z_A Z_M - Z_B Z_L|$$

$$D_2 = \left| \frac{(\chi_M + \chi_N) \cdot Z_E}{\chi_A} - (Z_M + Z_N) - |Z_M - Z_N| \right|$$



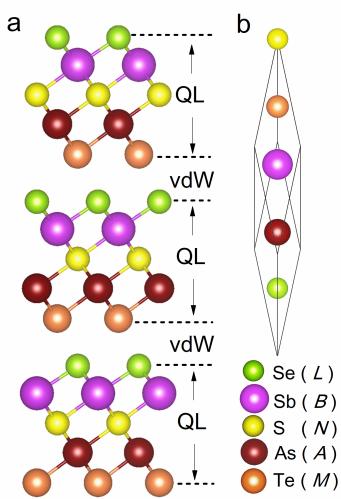
SISSO: predicting new tetradymite topological insulators

Prototype formula:

AB-LNM

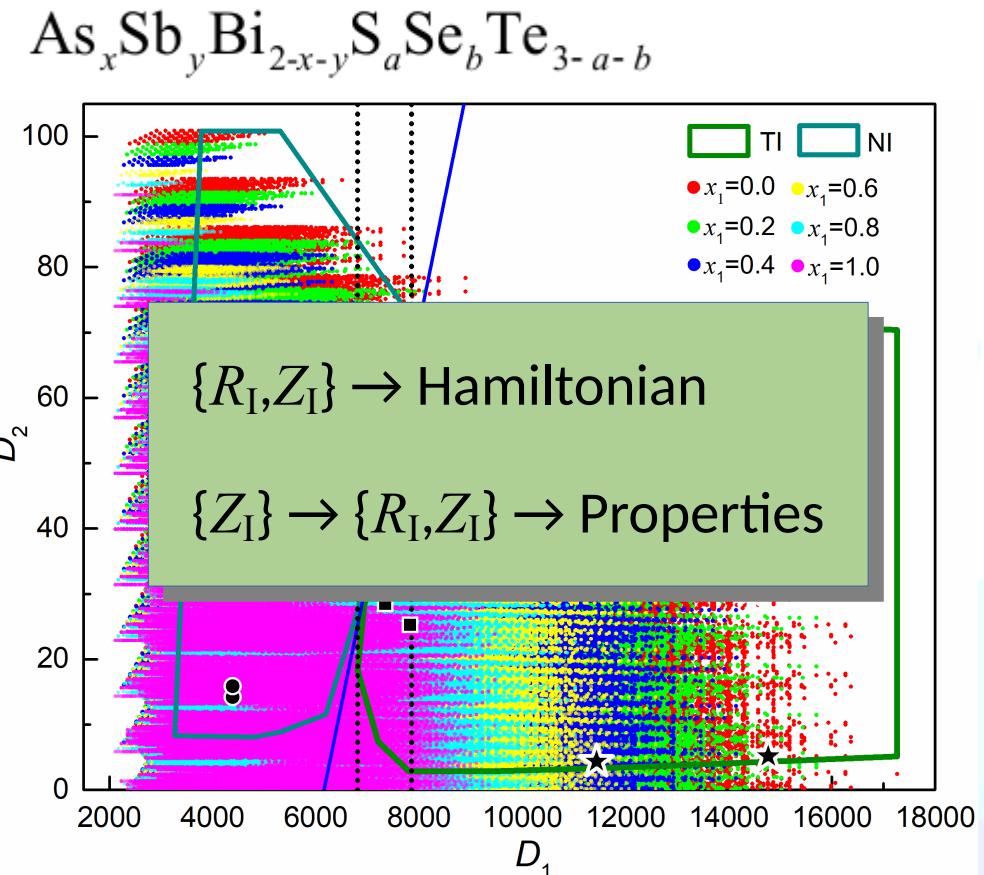
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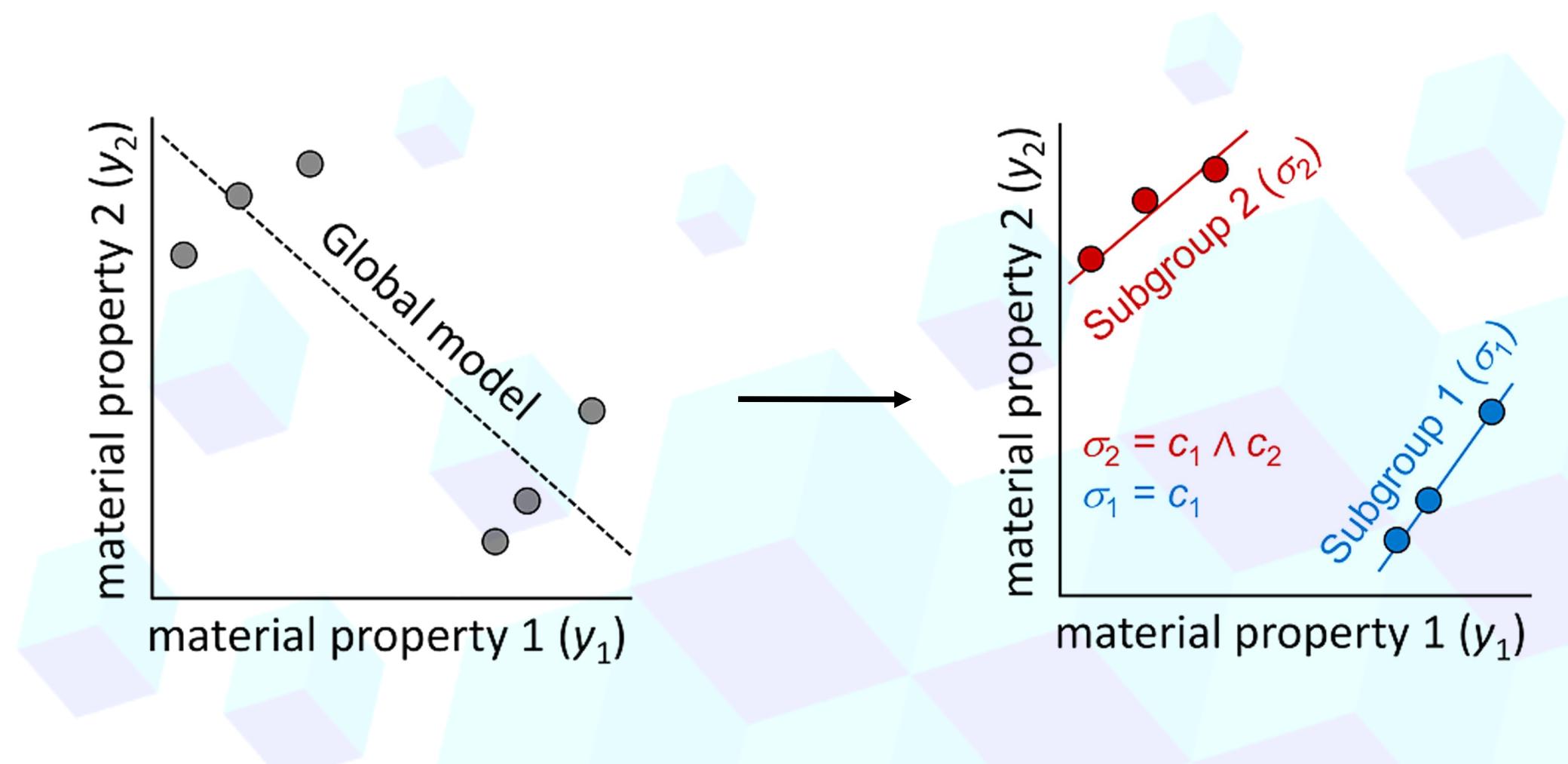
Compressed-sensing-based model identification (SISSO, and beyond): The context

$$\arg \min_{\mathbf{c}} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

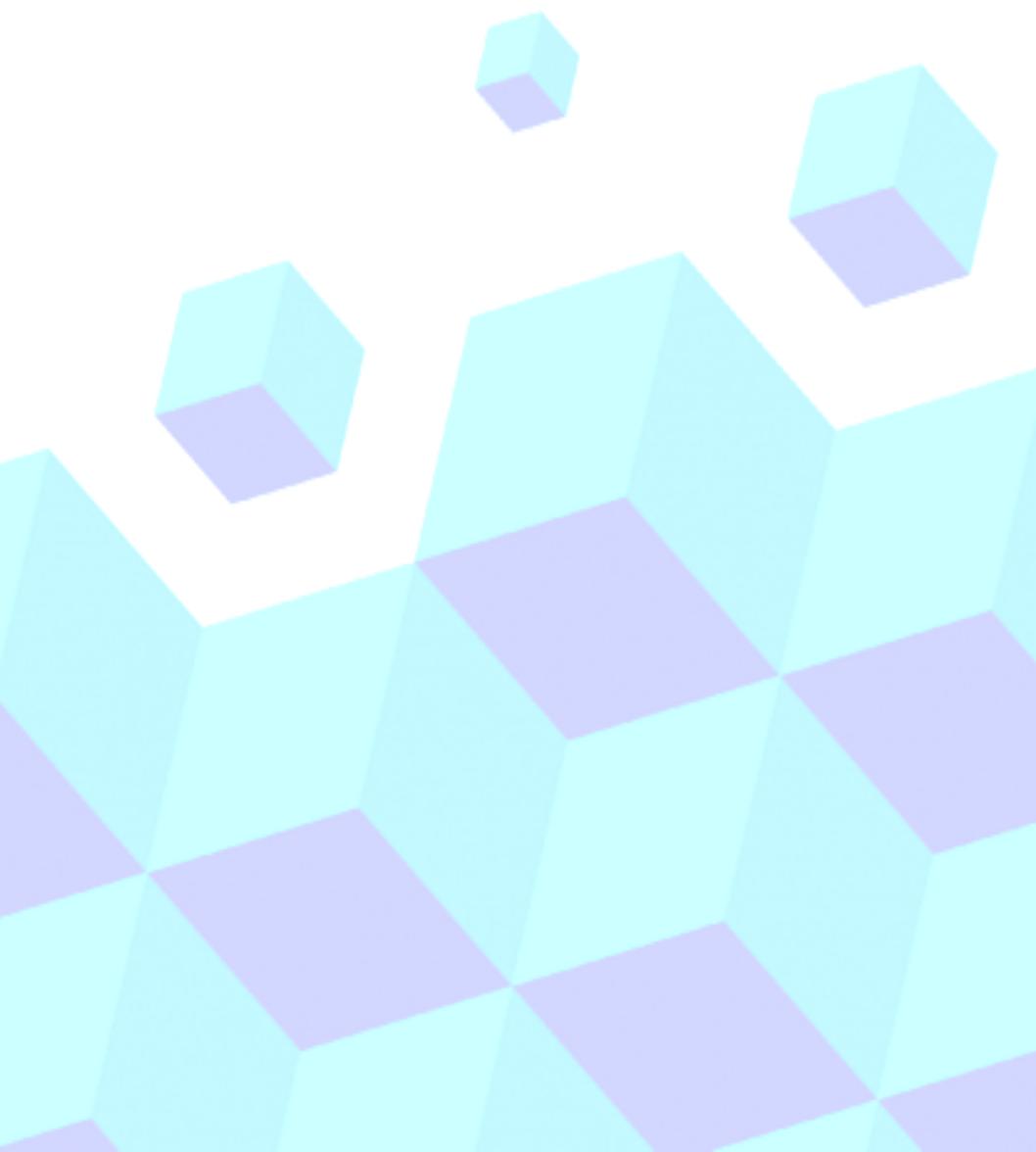
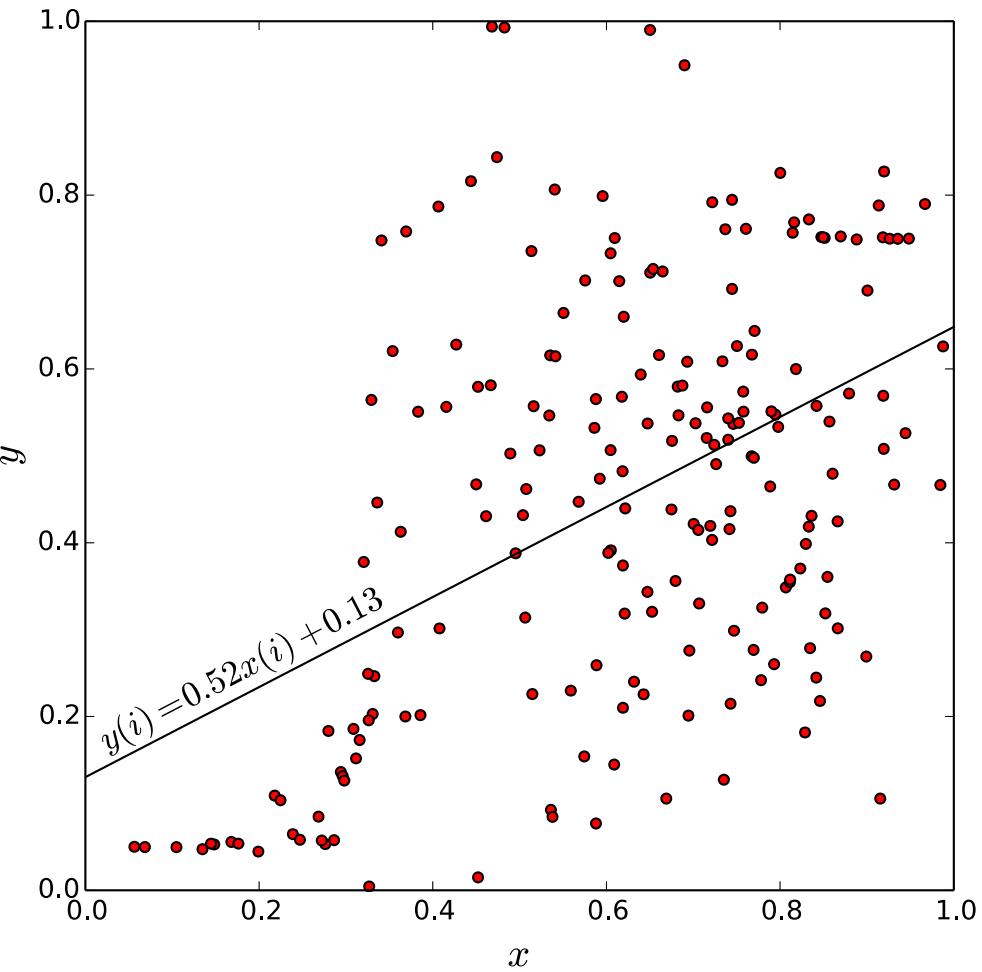
Compressed-sensing-based model identification:
Shares concepts with

- Regularized regression. But: Massive sparsification.
- Dimensionality reduction. But supervised, and yielding sparse, “inspectable” descriptors
- Feature/Basis-set selection/extraction. But: non-greedy solver.
- Symbolic regression. But: deterministic solver.

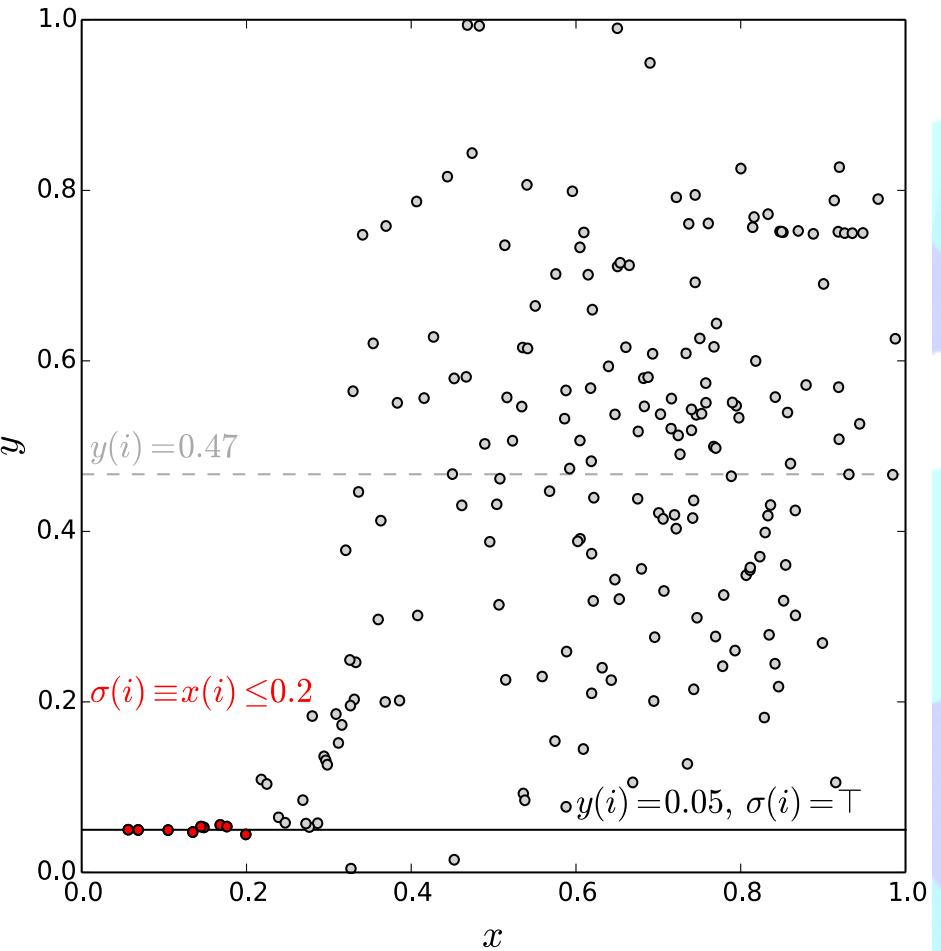
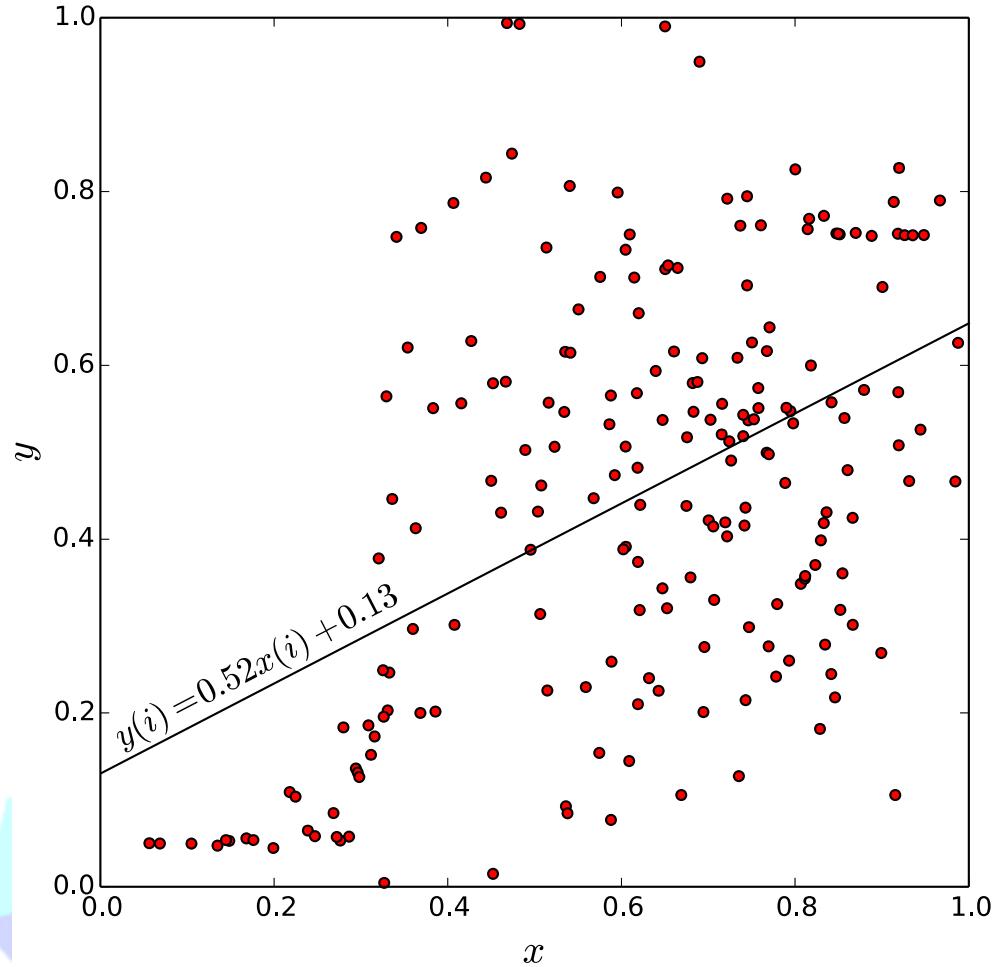
Subgroup discovery



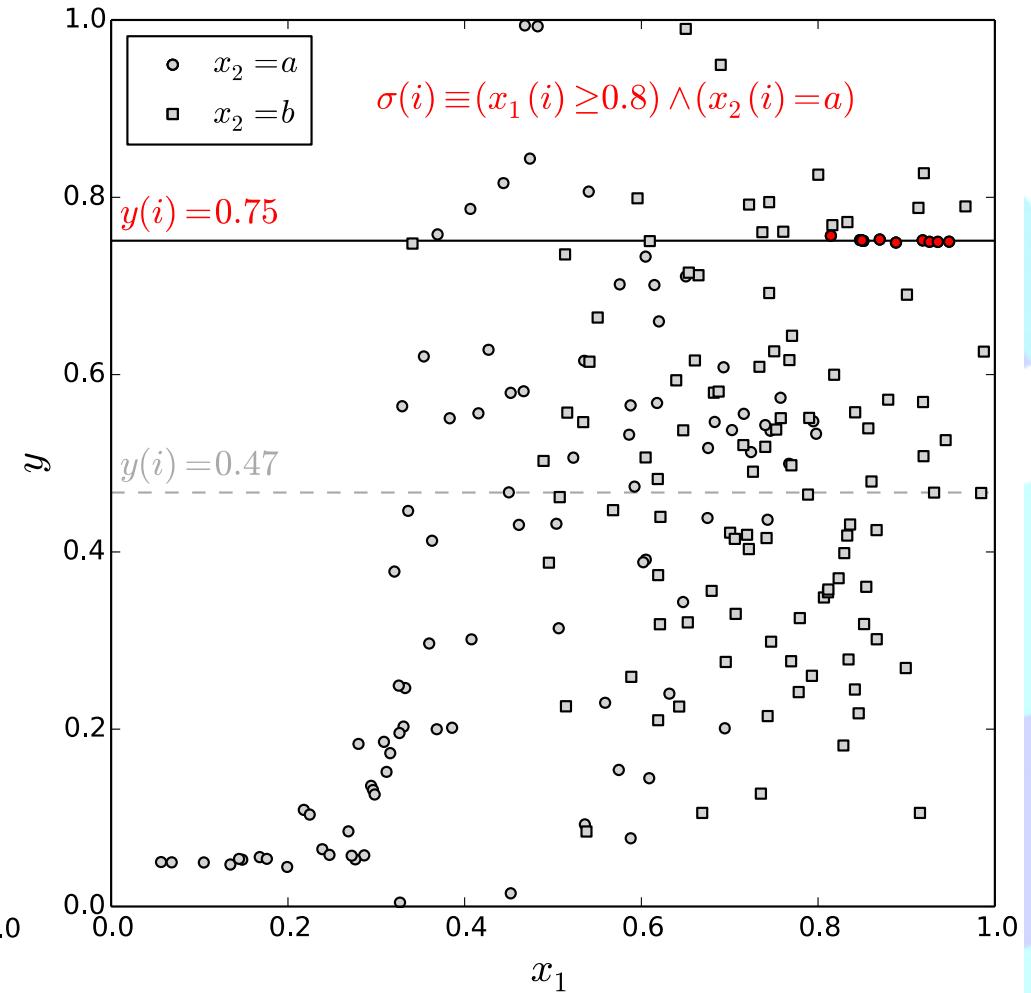
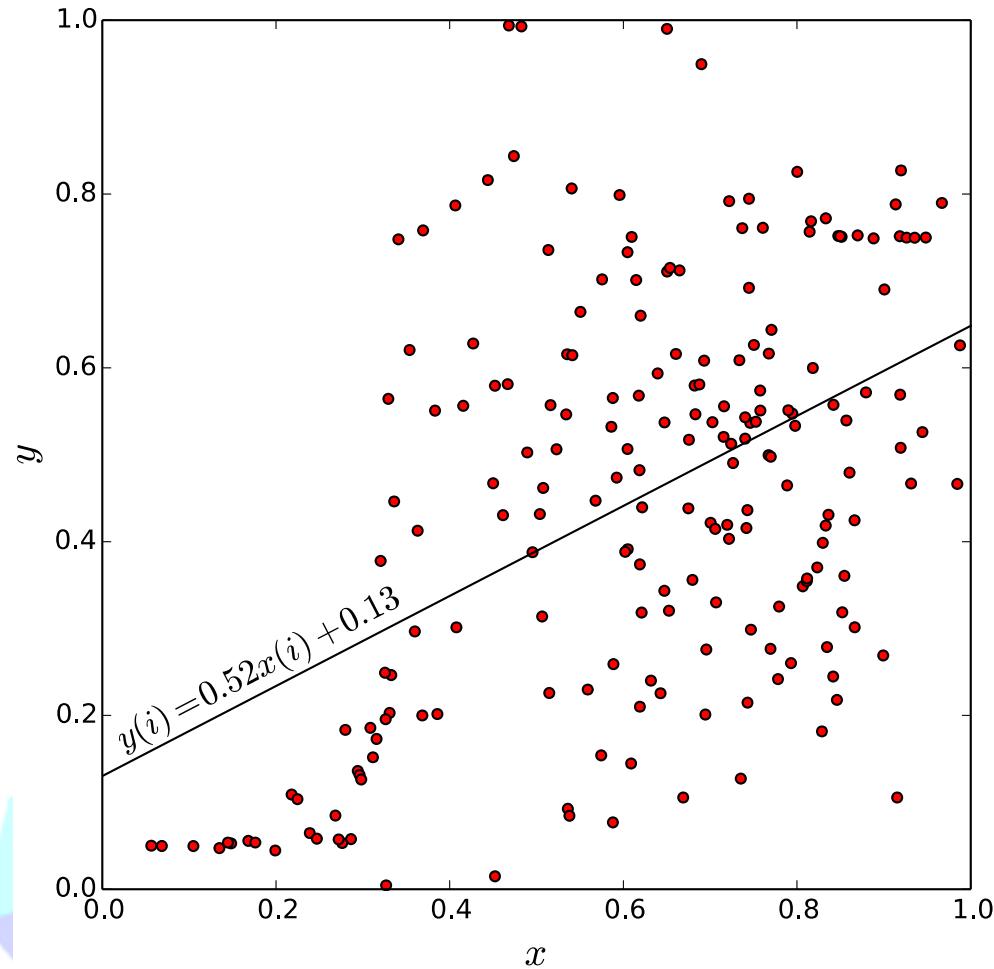
Subgroup discovery: finding descriptive statements about outstanding groups



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Subgroup discovery: finding descriptive statements about outstanding groups

Ingredients:

Population $P = \{1, \dots, n\}$

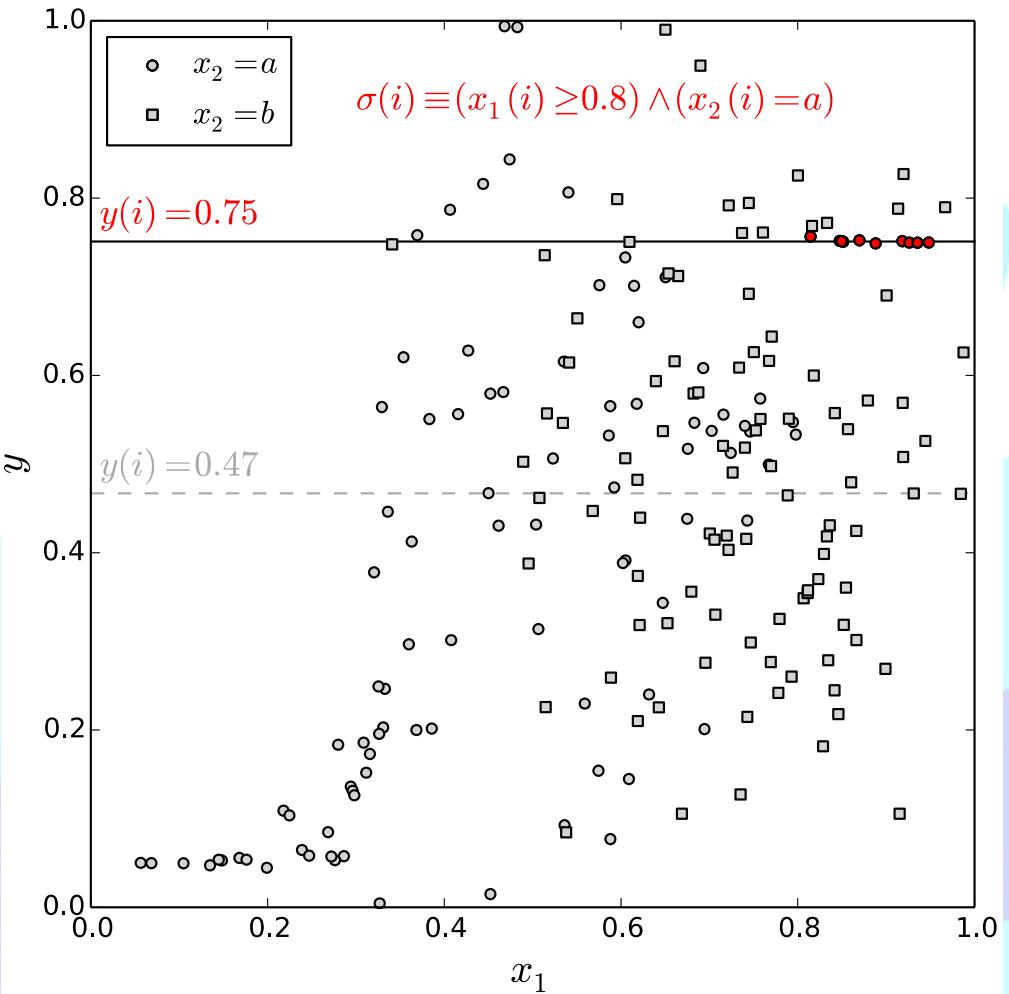
Target Variable $y: P \rightarrow Y$

Description variables $x_j: P \rightarrow X_j$

Basic propositions $\Pi = \{\pi_1, \dots, \pi_k\}$

Objective functions: f

{All possible subgroups of $P\} \rightarrow \mathbb{R}$



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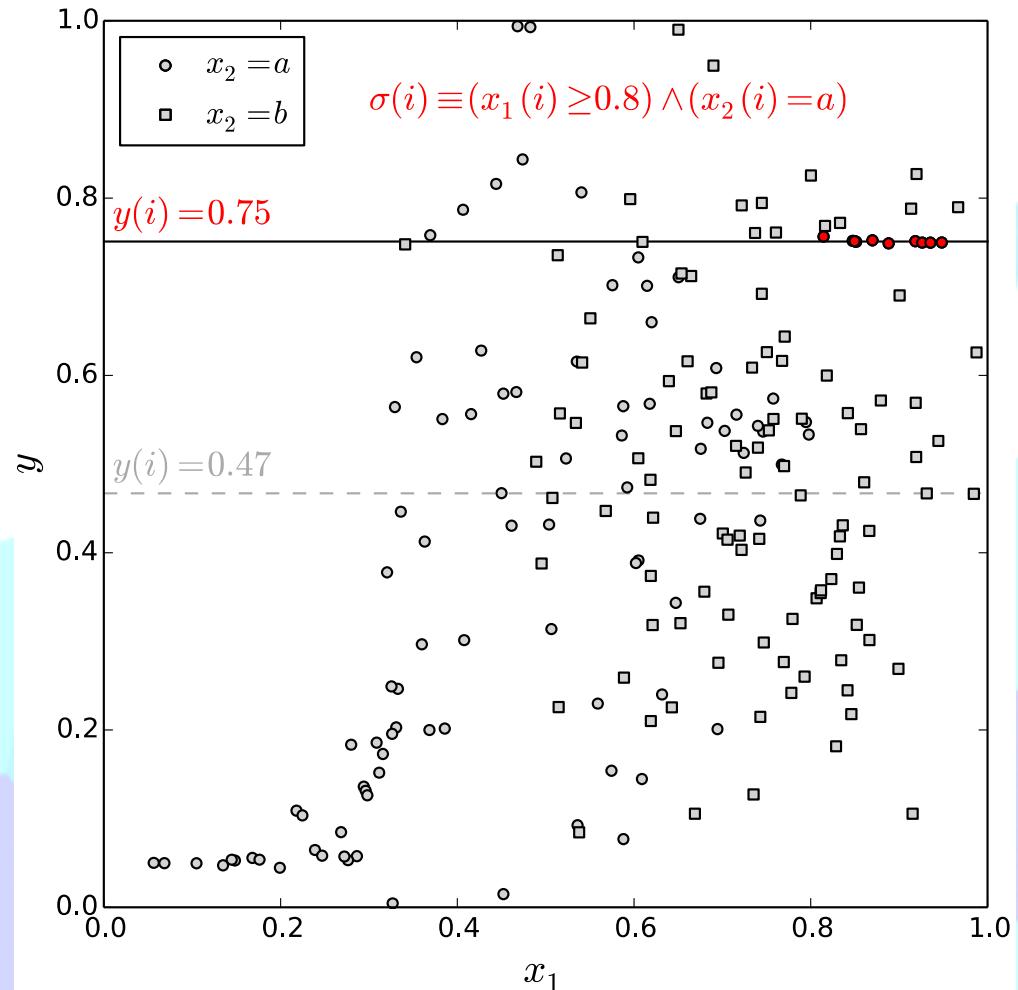
Task:

Finding $\sigma(i) = \pi_1(i) \wedge \dots \wedge \pi_m(i)$

For which $f(\sigma(i)) = \max$

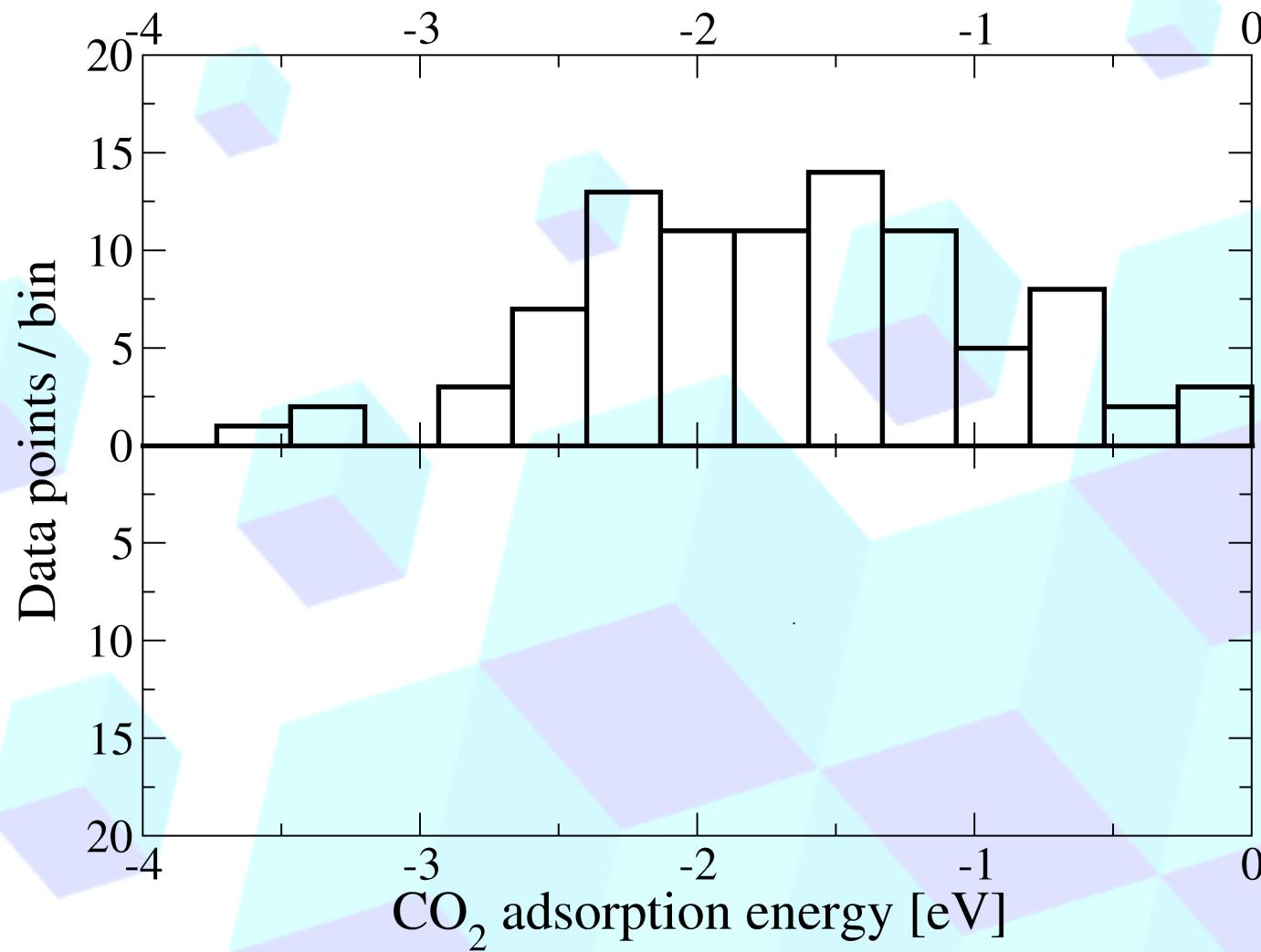
Typical form of f :

“Size of subgroup” \times “Reduction of variance of Y compared to the whole population”



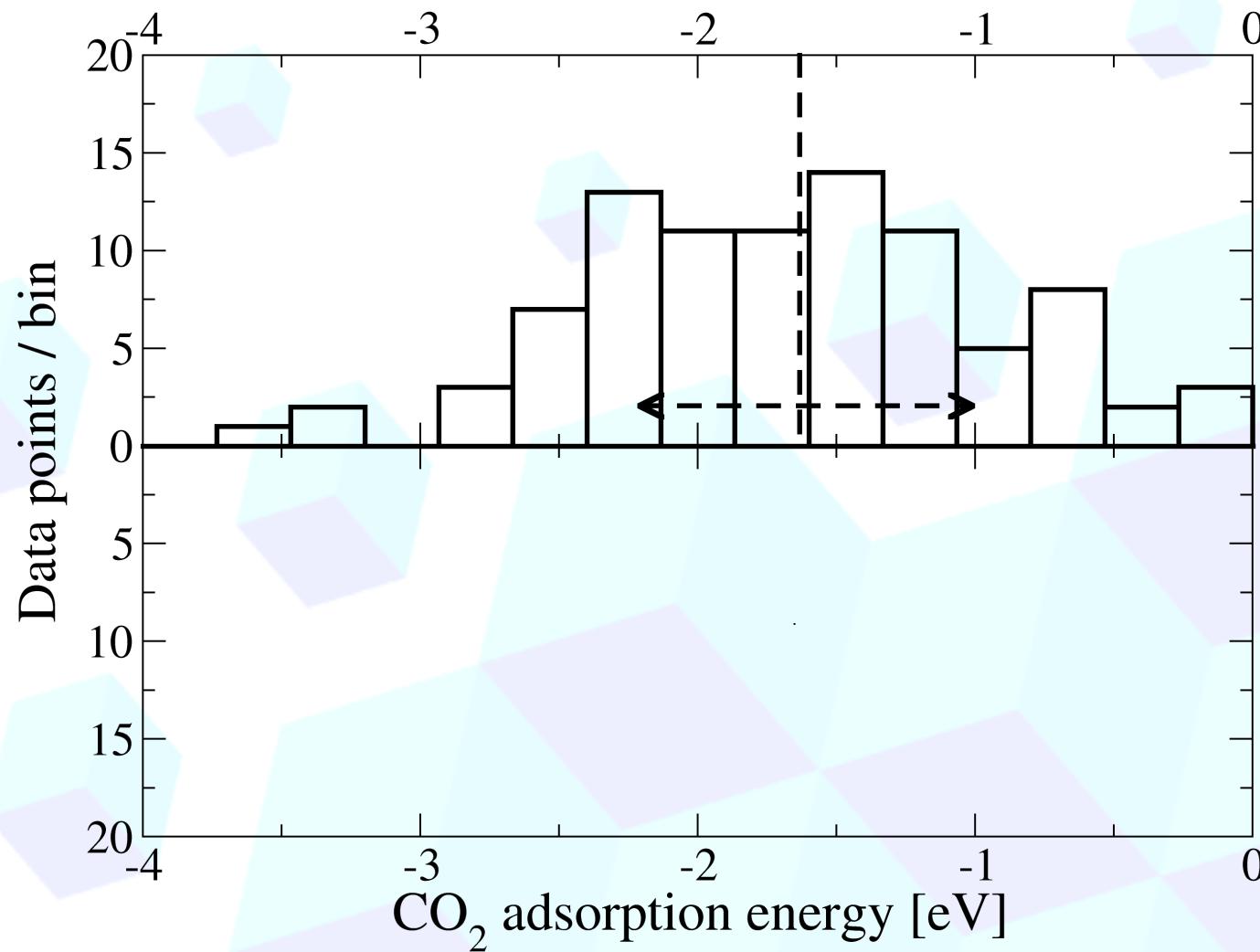
Subgroup discovery in practice

Distribution of adsorption energies of CO₂ on different surfaces of several metal-oxides



Subgroup discovery in practice

Distribution of adsorption energies of CO₂ on different surfaces of several metal-oxides



Subgroup discovery in practice

$$\operatorname{argmax}_{SG \subset P} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right) |\text{med}(SG) - \text{med}(P)|$$

Subgroup discovery in practice

$$\underset{SG \subset P}{\operatorname{argmax}} U = \frac{\#SG}{\#P} \left(1 - \frac{\operatorname{mad}(SG)}{\operatorname{mad}(P)} \right) |\operatorname{med}(SG) - \operatorname{med}(P)|$$

Size of subgroup SG

Size of full set P

Subgroup discovery in practice

$$\operatorname{argmax}_{SG \subset P} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right) |\text{med}(SG) - \text{med}(P)|$$

Mean absolute deviation
from the median
(spread of distribution)

Size of subgroup SG

Size of full set P

Subgroup discovery in practice

$$\operatorname{argmax}_{SG \subset P} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right) |\text{med}(SG) - \text{med}(P)|$$

Mean absolute deviation
from the median
(spread of distribution)

Median of the distribution

Size of subgroup SG

Size of full set P

The diagram illustrates the components of the U-measure formula. It shows arrows pointing from the labels 'Size of subgroup SG' and 'Size of full set P' to the terms $\#SG/\#P$ and $|\text{med}(SG) - \text{med}(P)|$ respectively. Another arrow points from the label 'Mean absolute deviation from the median (spread of distribution)' to the term $\text{mad}(SG)/\text{mad}(P)$.

Subgroup discovery in practice

$$\operatorname{argmax}_{SG \subset P} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right)$$

Annotations for the equation:

- Size of subgroup SG (points to $\#SG$)
- Size of full set P (points to $\#P$)
- Mean absolute deviation from the median (spread of distribution) (points to $\text{mad}(SG)$ and $\text{mad}(P)$)
- Maximize median shift (points to $| \text{med}(SG) - \text{med}(P) |$)
- Median of the distribution (points to $\text{med}(SG)$ and $\text{med}(P)$)
- Minimize relative spread of SG (points to the entire fraction $\frac{\text{mad}(SG)}{\text{mad}(P)}$)

Subgroup discovery in practice

$$\operatorname{argmax}_{SG \subset P} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right)$$

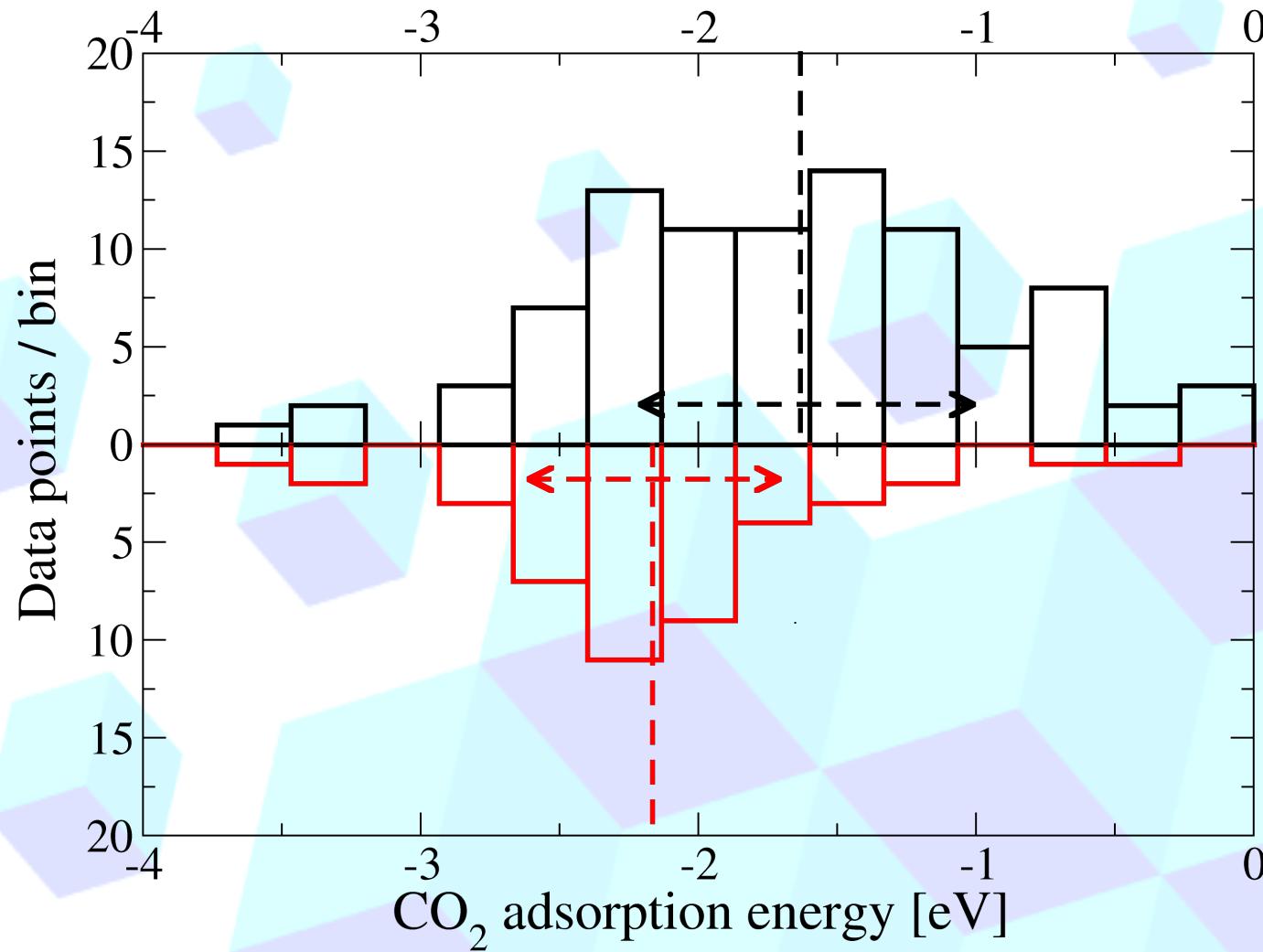
Annotations for the equation:

- Size of subgroup SG** : Points to $\#SG$.
- Size of full set P** : Points to $\#P$.
- Mean absolute deviation from the median (spread of distribution)**: Points to $\text{mad}(SG) / \text{mad}(P)$.
- Maximize median shift**: Points to $|\text{med}(SG) - \text{med}(P)|$.
- Median of the distribution**: Points to $\text{med}(SG)$.
- Minimize relative spread of SG** : Points to the entire term $\text{mad}(SG) / \text{mad}(P)$.

SG is described by a selector,
a conjunction of statements ($s_1 \wedge s_2 \wedge \dots$)
about a list of given features e.g.,
 s_1 = surface energy larger than ... ,
 s_2 = p -band center of surface O less than ...

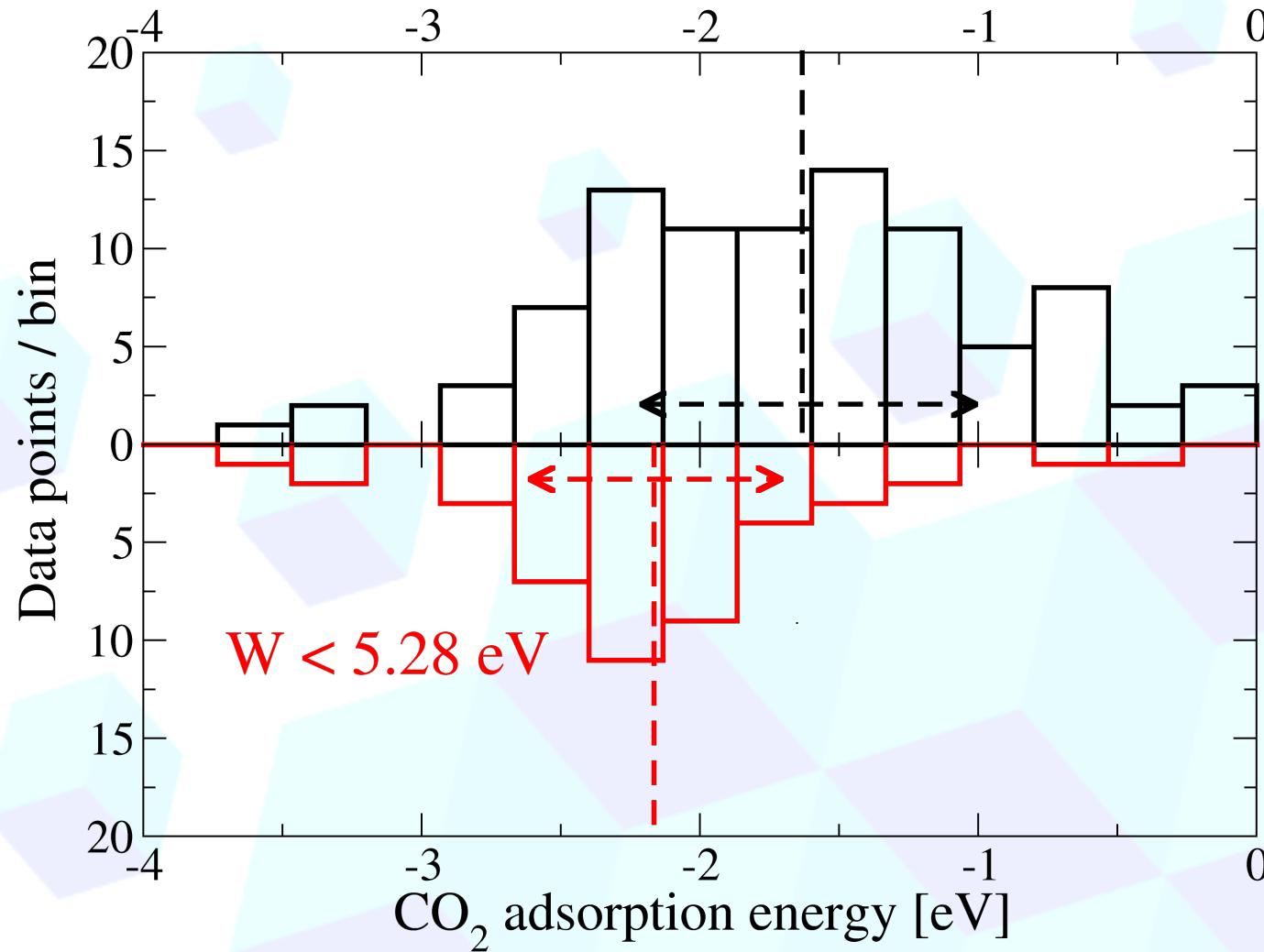
Subgroup discovery in practice

$$\underset{SG \subset P}{\operatorname{argmax}} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right) |\text{med}(SG) - \text{med}(P)|$$

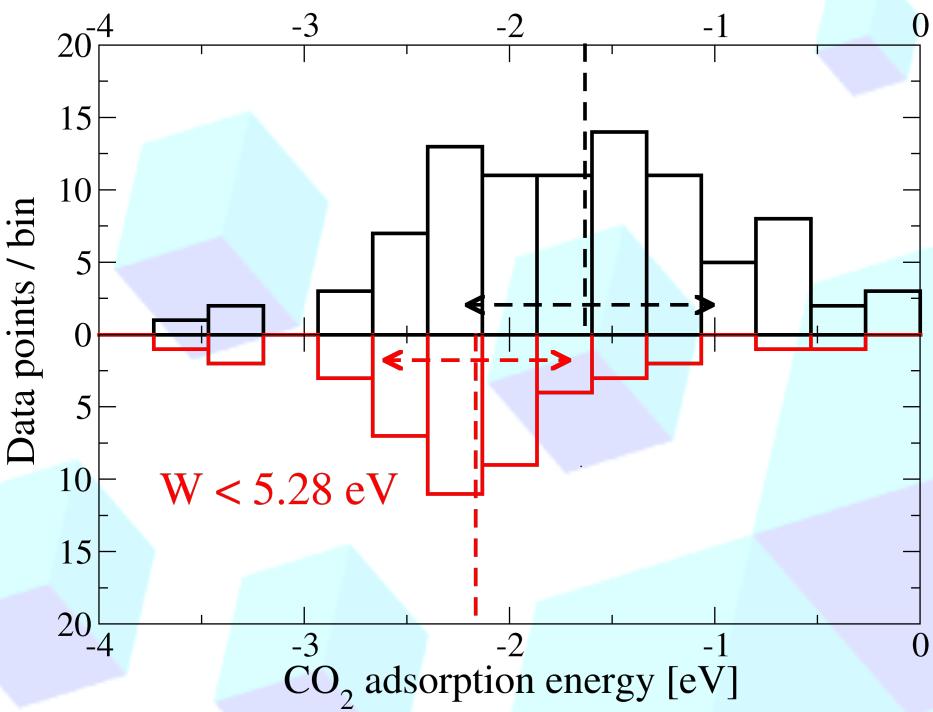


Subgroup discovery in practice

$$\operatorname{argmax}_{SG \subset P} U = \frac{\#SG}{\#P} \left(1 - \frac{\text{mad}(SG)}{\text{mad}(P)} \right) |\text{med}(SG) - \text{med}(P)|$$



Subgroup discovery in practice



The (SIS)O model for the discovered subgroup

- is more accurate than the global model
- has a different descriptor due to different physics.

Small work function:
Surfaces with dominantly ionic character

Acknowledgements

Compressed sensing, SISSO, and metal/insulator proof of concepts

Jan Vybiral, Runhai Ouyang, Emre Ahmetcik, Stefano Curtarolo, Sergey Levchenko, Claudia Draxl

Application of SISSO to perovskites

Christopher J. Bartel, Christopher Sutton, Bryan R. Goldsmith, Runhai Ouyang, Charles B. Musgrave

Application of SISSO to topological insulators

Guohua Cao, Runhai Ouyang, Zizhen Zhou, Huijun Liu, Christian Carbogno, Zhenyu Zhangave

Transparent conducting oxide: NOMAD-kaggle competition

Christopher Sutton, Angelo Ziletti, Claudia Draxl, Daan Frenkel, Kristian Thygesen, Samuel Kaski, Bernhard Schölkopf

Subgroup Discovery and application to CO₂ adsorption

Mario Boley, Jilles Vreeken, Aleksei Mazheika, Sergey Levchenko

All the above
Matthias Scheffler



NOMAD has received funding from the European Union's Horizon 2020 research and innovation programme under grant agreement No. 676580.



Acknowledgements

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Guohua Cao, Runhai Ouyang, Zizhen Zhou, Huijun Liu, Christian Carbogno, Zhenyu Zhangave

Tutorial (jupyter notebook)

On symbolic + regularized regression (from linear regression to SISSO)

Ask me (luca@fhi-berlin.mpg.de)
for user ID and password

All the above
Matthias Scheffler



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