Mechanical Behavior of Materials Lec.07-12

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This recitation class will cover lecture 7 and lecture $10\sim12$, the topics include:

- Phase Transformation & Kinetics (Lec.7)
- Basic Mechanical Testing Methods (Lec.10)
- Elastic Stress-Strain Behavior (Lec.11)
- Plane Stresses, Mohr's Circle & 3D Stresses (Lec.12)

The contents of Survey of Engineering Materials (Lec.8&9) will be covered in the RC for midterm review.

Agenda

- Phase Transformation & Kinetics
- 2 Basic Mechanical Testing Methods
- 3 Elastic Stress-Strain Behavior
- 4 Plane Stresses, Mohr's Circle & 3D Stresses

Process of Phase Transformation

The process of a phase Transformation is broken down into two distinct stages:

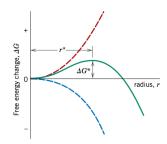
- Nucleation: the appearance of nuclei of the new phase, which are capable of growing.
 - Homogeneous: nuclei of the new phase form uniformly throughout the parent phase.
 - Heterogeneous: nuclei form preferentially at structural inhomogeneities such as grain boundaries, dislocations, and interfaces.
- **Growth:** the nuclei increase in size, which results in the disappearance of the parent phase.
- The Transformation reaches completion when equilibrium fraction of the new phase is attained.

Homogeneous Nucleation

For solidification, the driving force for the phase transformation is as follows:

$$\Delta G = \frac{4}{3}\pi r^3 \Delta G_v + 4\pi r^2 \gamma$$

- ΔG_v : the volume free energy difference between solid and liquid; $\Delta G_v < 0$ when $T < T_m$.
- γ : the solid-liquid surface energy; $\gamma > 0$.



- Growth will continue when $r > r^*$.
- Otherwise the nuclei will shrink and redissolve.
- Activation free energy (energy barrier) ΔG^* is required for the formation of a stable nucleus.

Homogeneous Nucleation

It can be shown that ΔG_v is a function of temperature as:

$$\Delta G_{v} = \frac{\Delta H_{f}(T_{m} - T)}{T_{m}}$$

where ΔH_f is the latent heat of fusion. Then the critical radius is given by:

$$r^* = -\frac{2\gamma}{\Delta G_v} = \left(-\frac{2\gamma T_m}{\Delta H_f}\right) \left(\frac{1}{T_m - T}\right)$$

The activation free energy is given by:

$$\Delta \textit{G}^* = rac{16\pi \gamma^3}{3\Delta \textit{G}_{\textit{v}}^2} = \left(rac{16\pi \gamma^3 \textit{T}_m^2}{3\Delta \textit{H}_f^2}
ight)rac{1}{\left(\textit{T}_m - \textit{T}
ight)^2}$$

Heterogeneous Nucleation

The critical radius for heterogeneous nucleation is given by:

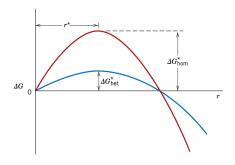
$$r_{\mathsf{het}}^* = -\frac{2\gamma_{\mathsf{SL}}}{\Delta G_{\mathsf{v}}} = r_{\mathsf{hom}}^*$$

The activation free energy is given by:

$$\Delta G_{\mathsf{het}}^* = \left(rac{16\pi\gamma_{\mathsf{SL}}^3}{3\Delta G_{\mathsf{v}}^2}
ight) S(heta) = \Delta G_{\mathsf{hom}}^* S(heta)$$

where

$$S(\theta) = \frac{2 - 3\cos\theta + \cos^3\theta}{4}$$



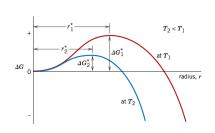
- The critical radius is the same as that for homogeneous nucleation.
- The activation energy is lowered by factor $S(\theta)$ because the surface energy is reduced.

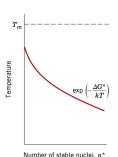
Effect of Temperature

Both the critical radius r^* and the activation free energy ΔG^* decrease as temperature T decreases. The number of stable nuclei n^* is a function of temperature:

$$n^* = K_1 \exp\left(-\frac{\Delta G^*}{kT}\right)$$

where K_1 is a constant related to total number of nuclei.





T just below TE
Nucleation rate low
Growth rate high
Large crystal
Higher diffusivity

T moderately below
Nucleation rate med .
Growth rate med .
Medium crystal

pearlite

Effect of Time

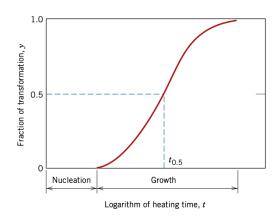
For a fixed temperature, the fraction of transformation *y* is a function of time given by the **Avrami equation**:

$$y = 1 - \exp(-kt^n)$$

where k and n are time-independent constants. By convention, the rate of a transformation is defined as

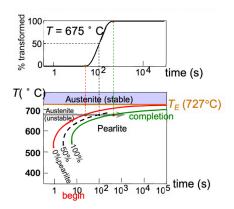
$$\mathsf{rate} = \frac{1}{t_0}$$

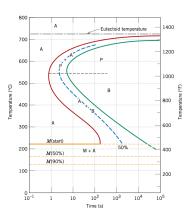
where $t_{0.5}$ is the time required for y=0.5. The rate of transformation is a function of temperature and the Avrami equation can be used to determine the rate of transformation.



TTT Diagram

- TTT stands for **Time-Temperature-Transformation**.
- TTT diagrams provide the following information:
 - Nature and type of transformation.
 - Stability of phases under isothermal transformation conditions.
 - Temperature or time required to start or finish transformation and rate of transformation.





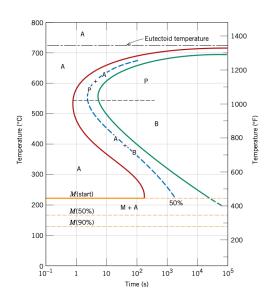
Microstructures in Fe-Fe₃C System

- Pearlite: alternating ferrite+cementite layers.
 - High T, both C and Fe diffuse.
 - Coarse pearlite: high diffusion rate and low nucleation rate, smaller ΔT
 - Fine pearlite: low diffusion rate and high nucleation rate, larger ΔT
- Bainite: non layered ferrite+cementite.
 - Moderate T, C diffuses.
 - Note that bainite forms directly from austenite and not from pearlite.
- Martensite: ferrite with supersaturated carbon.
 - Low T, C and Fe do not diffuse but just slight atomic displacement.

Example

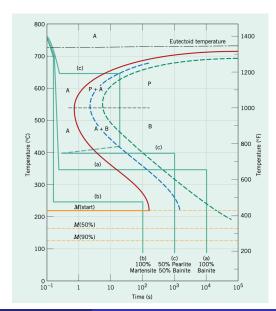
Specify the nature of the final microstructure of a small specimen of eutectoid composition that has been subjected to the following time-temperature treatments. In each case assume that the specimen begins at $760^{\circ}C$ and that it has been held at this temperature long enough to have achieved a complete and homogeneous austenitic structure.

- (a) Rapidly cool to 350° C, hold for 10^{4} s, and quench to room temperature.
- (b) Rapidly cool to 250° C, hold for 10^{2} s, and quench to room temperature.
- (c) Rapidly cool to 650° C, hold for 20 s, rapidly cool to 400° C, hold for 10^{3} s, and quench to room temperature.



Example

- (a) At $350^{\circ}C$, reaction takes about 10s to start and reach final state (bainite) after about $500s \Rightarrow 100\%$ bainite.
- (b) At 250° C, reaction takes more than 100s to reach final state, but the process only takes $100s \Rightarrow 100\%$ austenite. Austenite then cools down into martensite $\Rightarrow 100\%$ martensite.
 - c) At $650^{\circ}C$, after 20 s, the reaction stops on the blue line $\Rightarrow 50\%$ austenite & 50% pearlite. At $400^{\circ}C$, all remaining austenite changes into bainite after $100s \Rightarrow 50\%$ bainite & 50% pearlite.

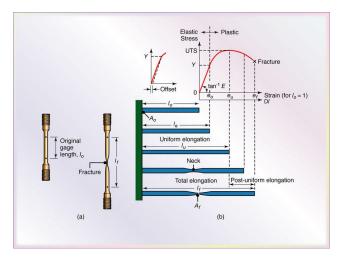


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Stress-Strain Curve

Stress-strain curve can be used to determine the following mechanical properties: (1) ultimate tensile strength; (2) yield strength; (3) ductility; (4) toughness; (5) modulus of elasticity.



Strength

Ultimate Tensile Strength

The ultimate tensile strength σ_u is the highest engineering stress on the stress-strain curve.

$$\sigma_u = \frac{P_{\mathsf{max}}}{A_i}$$

For brittle materials, $\sigma_u = \sigma_f$, for ductile materials, $\sigma_u > \sigma_f$, where σ_f is the stress at fracture.

Yield Strength

The yield strength σ_y is maximum stress that can be developed in a material without causing plastic deformation.

Ductility

Ductility

Ductility is the ability of materials to accommodate inelastic deformation without breaking.

Ductility can be measured by:

• Engineering strain at fracture:

$$\varepsilon_f = \frac{L_f - L_0}{L_0}$$

• Engineering strain after fracture:

$$\varepsilon_{pf} = \frac{L_{pf} - L_i}{L_i} = \varepsilon_f - \frac{\sigma_f}{F}$$

Percent reduction in area:

$$%RA = 100 \frac{A_i - A_f}{A_i} = 100 \frac{d_i^2 - d_f^2}{d_i^2}$$

Tensile Toughness

Tensile Toughness

The area under the entire engineering stress-strain curve up to fracture is called the tensile toughness, u_f , which is a measure of the ability of the material to absorb energy without fracture.

$$u_f = \int_0^{\varepsilon_f} \sigma \mathrm{d}\varepsilon$$

For brittle materials:

$$u_f \approx \frac{2}{3} \sigma_f \varepsilon_f$$

For ductile materials:

$$u_f \approx \varepsilon_f \left(\frac{\sigma_y + \sigma_u}{2} \right)$$

True Stress and Strain

True Stress

True stress is the axial force P divided by the current cross-sectional area A instead of A_i :

$$\tilde{\sigma} = \frac{P}{A} = \sigma \frac{A_i}{A}$$

True Strain

The true strain is given as:

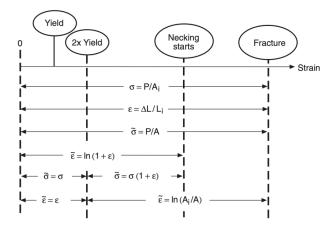
$$ilde{arepsilon} = \operatorname{In}\left(rac{L}{L_i}
ight) = \operatorname{In}\left(1+arepsilon
ight)$$

Constant Volume Assumption

With constant volume assumption $A_iL_i = AL$ (invalid for small strains), we have

$$ilde{\sigma} = \sigma(1+arepsilon) \quad ext{and} \quad ilde{arepsilon} = \ln rac{A_i}{A}$$

Limitations on True Stress-Strain Calculation



Once necking starts at the engineering ultimate stress point, the engineering strain becomes merely an average over a region of nonuniform deformation. Hence, it does not represent the maximum strain and becomes unsuitable for calculating true stresses and strains.

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Hooke's Law

For homogeneous and isotropic materials,

$$\begin{bmatrix} \varepsilon_{x} \\ \varepsilon_{y} \\ \varepsilon_{z} \\ \gamma_{xy} \\ \gamma_{yz} \\ \gamma_{zx} \end{bmatrix} = \begin{bmatrix} \frac{1}{E} & -\frac{\nu}{E} & -\frac{\nu}{E} & 0 & 0 & 0 \\ -\frac{\nu}{E} & \frac{1}{E} & -\frac{\nu}{E} & 0 & 0 & 0 \\ -\frac{\nu}{E} & -\frac{\nu}{E} & \frac{1}{E} & 0 & 0 & 0 \\ 0 & 0 & 0 & \frac{1}{G} & 0 & 0 \\ 0 & 0 & 0 & 0 & \frac{1}{G} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{1}{G} \end{bmatrix} \begin{bmatrix} \sigma_{x} \\ \sigma_{y} \\ \sigma_{z} \\ \tau_{xy} \\ \tau_{yz} \\ \tau_{zx} \end{bmatrix} + \begin{bmatrix} \varepsilon_{T} \\ \varepsilon_{T} \\ 0 \\ 0 \\ 0 \end{bmatrix}$$

- Volumetric strain: $\varepsilon_{v} = \varepsilon_{x} + \varepsilon_{y} + \varepsilon_{z} = \frac{(1-2v)(\sigma_{x}+\sigma_{y}+\sigma_{z})}{E}$
- Thermal strain: $\varepsilon_T = \alpha \Delta T$
- Three elastic constants, 2 independent: $G = \frac{E}{2(1+v)}$

Example

Strains are measured on the surface of an aluminum alloy part as follows:

$$\varepsilon_{x} = 400 \times 10^{-6}, \quad \varepsilon_{y} = 1000 \times 10^{-6}, \quad \gamma_{xy} = 800 \times 10^{-6}$$

Estimate the in-plane stress components σ_x , σ_y and τ_{xy} , and also the strain ε_z normal to the surface. Note that for aluminum alloy,

$$E = 70.3 \times 10^3 \text{ MPa}, \quad \nu = 0.345$$

(Assume that $\sigma_z = \tau_{yz} = \tau_{zx} = 0$.)

Example

According to Hooke's law,

$$\begin{cases} \varepsilon_{x} = \frac{1}{E}\sigma_{x} - \frac{\nu}{E}\sigma_{y} = 400 \times 10^{-6} \\ \varepsilon_{y} = \frac{1}{E}\sigma_{y} - \frac{\nu}{E}\sigma_{x} = 1000 \times 10^{-6} \\ \gamma_{xy} = \frac{1}{G}\tau_{xy} = \frac{2(1+\nu)}{E}\tau_{xy} = 800 \times 10^{-6} \end{cases}$$

Therefore.

$$\begin{cases} \sigma_x = 59.4 \text{ MPa} \\ \sigma_y = 90.8 \text{ MPa} \\ \tau_{xy} = 20.9 \text{ MPa} \end{cases}$$

Then

$$\varepsilon_z = -\frac{\nu}{F}(\sigma_x + \sigma_y) = -7.37 \times 10^{-4}$$

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Ductile vs. Brittle Failure

- Ductile material: if maximum principle shear stress τ_{max} is larger than critical shear $\tau_y = \sigma_y/2$, the material yields.
- Brittle material: if maximum principle normal stress σ_{max} is larger than ultimate tensile strength σ_u , the material fractures.

Plane Stress Transformation

Plane stress when:

$$\sigma_z = \tau_{zx} = \tau_{yz} = 0$$

- eta is defined as positive in counterclockwise (CCW) rotation direction for coordinate systems.
- The new stresses are given by:

$$\sigma_{x'} = \frac{\sigma_x + \sigma_y}{2} + \frac{\sigma_x - \sigma_y}{2} \cos 2\theta + \tau_{xy} \sin 2\theta$$

$$\sigma_{y'} = \frac{\sigma_x + \sigma_y}{2} - \frac{\sigma_x - \sigma_y}{2} \cos 2\theta - \tau_{xy} \sin 2\theta$$

$$\tau_{x'y'} = -\frac{\sigma_x - \sigma_y}{2} \sin 2\theta + \tau_{xy} \cos 2\theta$$

Principle Stresses

Principle normal stresses:

$$\sigma_1, \sigma_2 = rac{\sigma_{\mathsf{x}} + \sigma_{\mathsf{y}}}{2} \pm \sqrt{\left(rac{\sigma_{\mathsf{x}} - \sigma_{\mathsf{y}}}{2}
ight)^2 + au_{\mathsf{xy}}^2}$$
 $an 2 heta_n = rac{2 au_{\mathsf{xy}}}{\sigma_{\mathsf{x}} - \sigma_{\mathsf{y}}}$

Principle shear stress:

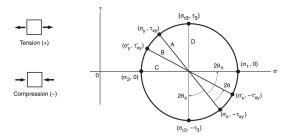
$$\begin{split} \tau_3 &= \sqrt{\left(\frac{\sigma_{\rm X} - \sigma_{\rm y}}{2}\right)^2 + \tau_{\rm xy}^2} = \frac{|\sigma_1 - \sigma_2|}{2} \\ \tan 2\theta_{\rm s} &= -\frac{\sigma_{\rm X} - \sigma_{\rm y}}{2\tau_{\rm xy}}, \quad \sigma_{\tau_3} = \frac{\sigma_{\rm X} + \sigma_{\rm y}}{2} = \frac{\sigma_1 + \sigma_2}{2} \end{split}$$

• $|\theta_s - \theta_n| = \frac{\pi}{4}$

Mohr's Circle

Equation of Mohr's circle:

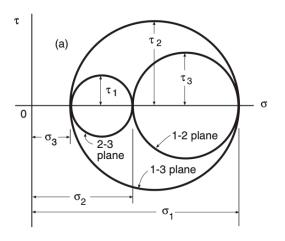
$$\left(\sigma - \frac{\sigma_{\mathsf{x}} + \sigma_{\mathsf{y}}}{2}\right)^{2} + \tau^{2} = \left(\frac{\sigma_{\mathsf{x}} - \sigma_{\mathsf{y}}}{2}\right)^{2} + \tau_{\mathsf{x}\mathsf{y}}^{2}$$



- A rotation of 2θ on Mohr's circle corresponds to a rotation of θ for coordinate system in real space.
- Clockwise (CW) shear stress: positive.
- Counter Clockwise (CCW) shear stresses: negative

CCW(-)

3D State of Stress Transformation



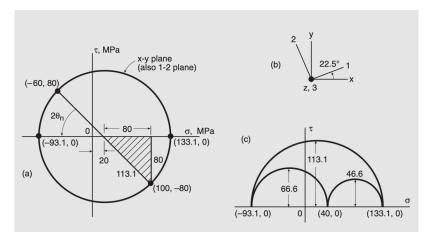
- Draw three Mohr's circles for xy, yz, and zx planes.
- Usually define $\sigma_1 > \sigma_2 > \sigma_3$
- The maximum shear stress for any plane in the material is the largest of the three principal shear stresses:

$$\tau_{\mathsf{max}} = \mathsf{max}(\tau_1, \tau_2, \tau_3) = \frac{\sigma_1 - \sigma_3}{2}$$

Example 6.3 in Dowling

For the following state of stress (in MPa), determine the principal normal stresses, the principal axes, and the principal shear stresses:

$$\sigma_{x} = 100, \sigma_{y} = -60, \sigma_{z} = 40, \tau_{xy} = 80, \tau_{yz} = \tau_{zx} = 0$$



Thank you!