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# Size effects on Curie temperature of ferroelectric particles

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**ABSTRACT** A model for the size dependence of the Curie temperature  $T_{\rm c}$  of perovskite ferroelectric particles without any free adjustable parameters has been developed. The model predicts that  $T_{\rm c}$  decreases with decreasing particle size. The predictions of the model are in agreement with experimental results for PbTiO<sub>3</sub> and BaTiO<sub>3</sub>.

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#### 1 Introduction

As ferroelectric device elements become smaller and smaller, with the dimensions of the ferroelectric material in the sub-micrometer range or even lower, the physical properties are size-dependent, and this must be taken into consideration in optimizing the properties [1, 2]. One of the most important properties is the size dependence of the Curie temperature. A quantitative model for  $T_c$  of PbTiO<sub>3</sub> particles has been established on the basis of mechanical and thermodynamic considerations using the Landau–Ginzburg–Devonshire (LGD) phenomenological theory [3]. The model prediction corresponds well to the experimental results for PbTiO<sub>3</sub> particles. However, the model cannot be extended to predict  $T_c$  of BaTiO<sub>3</sub> particles even though both have the same octahedral structure. Therefore, a general model to predict the size dependence of  $T_c$  is needed.

Recently, a thermodynamic model for the solid–liquid transition (a first-order transition) has been developed for crystals of finite size for metals [4], semiconductors [5], organics [6] and molecules [7]. The model has also been extended to the glass–liquid transition (a second-order transition) of polymers [8] and organics [9]. Since the model is suitable for predicting transitions of both the first and the second order and the Curie transition of a ferroelectric is also a first-order phase transition, finite-size effects on the Curie temperature  $T_{\rm c}$  of perovskite ferroelectric particles may be related to the corresponding effects on the melting temperature of nanocrystals. In this contribution, the size-dependent crystal–liquid transition model will be generalized to predict

finite-size effects on the Curie temperature of ferroelectrics. This extension of the model is found to be consistent with the experimental results for PbTiO<sub>3</sub> and BaTiO<sub>3</sub> particles.

## 2 Model

The size-dependent melting temperature  $T_{\rm m}(D)$  of a crystal with diameter D is given by [4–7]

$$\frac{T_{\rm m}(D)}{T_{\rm m0}} = \exp\left(-\frac{2S_{\rm m}}{3R}\frac{1}{D/D_0 - 1}\right),\tag{1}$$

where  $T_{\rm m0}$  is the melting temperature of the corresponding bulk crystal.  $S_{\rm m}$  denotes the bulk melting entropy and R is the ideal gas constant.  $D_0$  denotes a critical radius where a crystal particle and a liquid particle are indistinguishable.

To extend the above model to the size-dependent Curie transition temperature  $T_c$ ,  $S_m$  is substituted by the transition entropy  $S_0$  and  $D_0$  is defined as a critical particle size where the ferroelectric phase cannot exist or the Curie transition is absent.  $D_0$  is determined by [10]

$$D_0 = \sigma_{90} / \left(2k\alpha\beta^2\right),\tag{2}$$

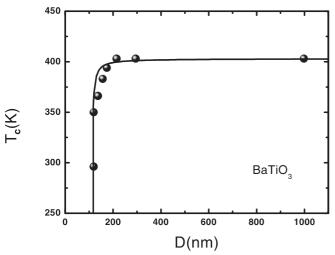
where  $\sigma_{90}$  denotes the density of 90° domain walls, k is a coefficient,  $\alpha$  is the elastic stiffness of the material and  $\beta = (c/a) - 1$ , with c and a being the lattice constants of the material. With the above definitions, (1) can be rewritten as

$$\frac{T_{\rm c}}{T_{\rm c0}} = \exp\left(-\frac{2S_0}{3R}\frac{1}{D/D_0 - 1}\right),\tag{3}$$

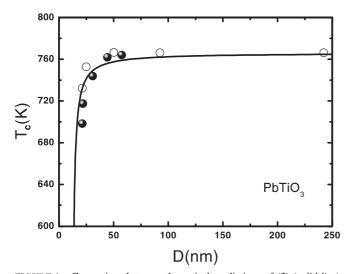
where  $T_{c0}$  is the bulk Curie temperature.

#### 3 Results and discussion

The  $T_c$  functions of BaTiO<sub>3</sub> and PbTiO<sub>3</sub> have been predicted using (3) and compared with experimental results. The predictions and experimental results are shown for BaTiO<sub>3</sub> and PbTiO<sub>3</sub> in Figs. 1 and 2, respectively. In the figures and in the predictions from (3),  $T_c$  decreases as D decreases. The predicted  $T_c$  functions of BaTiO<sub>3</sub> and PbTiO<sub>3</sub> are in agreement with the corresponding experimental results. Comparing Figs. 1 and 2, it is evident that the drops of  $T_c$  of the two compounds with decreasing D have different tendencies, since BaTiO<sub>3</sub> has a larger  $D_0$  value and a smaller  $S_0$ 



**FIGURE 1** Comparison between theoretical predictions of (3) (*solid line*) and experimental results ( $\bullet$ ) [11] for  $T_c$  of BaTiO<sub>3</sub> particles.  $S_0 = 0.1 \, \mathrm{J} \, \mathrm{g}$ -atom<sup>-1</sup> K<sup>-1</sup> [12],  $T_{c0} = 403 \, \mathrm{K}$  [13].  $D_0$  has been determined to be 40 nm [10]. However, the value of the tetragonality c/a in (2) decreases with decreasing particle size. Taking account of this consideration,  $D_0$  has been redetermined as 110 nm [11]



**FIGURE 2** Comparison between theoretical predictions of (3) (*solid line*) and experimental results ( $\bullet$  [14] and  $\circ$  [15]) for  $T_c$  of PbTiO<sub>3</sub> particles.  $S_0 = 0.46 \, \mathrm{J \, g\text{-}atom^{-1} \, K^{-1}}$  [16],  $T_{c0} = 765.1 \, \mathrm{K}$  [17].  $D_0 = 11.8 \, \mathrm{nm}$  has been determined from (2) using the following parameters:  $\sigma_{90} = 0.05 \, \mathrm{J/m^2}$  [18];  $\beta = (c/a) - 1 = 0.065$ , where  $c = 0.4156 \, \mathrm{nm}$  and  $a = 0.3902 \, \mathrm{nm}$  [17]; and  $k\alpha = 0.5 \times 10^9 \, \mathrm{N \, m^{-2}}$  [10]

value than has PbTiO<sub>3</sub>. The two effects of  $D_0$  and  $S_0$  on  $T_c$  lead to the result that the size effect in BaTiO<sub>3</sub> is weaker than that in PbTiO<sub>3</sub> in terms of (3). This difference could be due to the fact that BaTiO<sub>3</sub> has a higher domain wall energy density and a lower tetragonality c/a in comparison with PbTiO<sub>3</sub>. Thus, the Curie transition of BaTiO<sub>3</sub> can proceed easily, and a large-sized domain may be formed easily also.

Note that the Curie transition model [3], in terms of the LGD phenomenological theory, has the form

$$\frac{T_{\rm c}}{T_{\rm c0}} = 1 - \frac{C}{D},\tag{4}$$

where C is a material constant. Equation 4 in fact is the same as (3) when D is at least one order magnitude larger than  $D_0$ , owing to the mathematical relationship  $\exp(-x) \approx 1 - x$ . In this case, (3) can be rewritten as  $T_c/T_{c0} = 1 - C/D$ , where  $C = 2S_0D_0/3R$ . However, as D approaches  $D_0$ , (3) becomes more suitable for predicting the size-dependent Curie transition temperature. This could be the reason why (4) may predict the transition behavior of PbTiO<sub>3</sub>, with a smaller  $D_0$ , satisfactorily but fails to predict the behavior of BaTiO<sub>3</sub>, which has a larger  $D_0$ .

#### 4 Conclusion

Our model for the size dependence of melting has been generalized to describe the size dependence of the Curie temperature of PbTiO<sub>3</sub> and BaTiO<sub>3</sub> particles by making use of the common feature of a first-order transition in these problems. It was found that the results predicted by the model for the size-dependent Curie temperature of the above compounds were in agreement with the experimental results.

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### REFERENCES

- 1 R. Newnham: MRS Bulletin 22, 20 (1997)
- 2 J. Scott, C. Paz de Araujo: Science 246, 1400 (1989)
- 3 H. Huang, C.Q. Sun, P. Hing: J. Phys.: Condens. Matter 12, L127 (2000)
- 4 Z. Zhang, J.C. Li, Q. Jiang: J. Phys. D 33, 2653 (2000)
- 5 Z. Zhang, M. Zhao, Q. Jiang: Semicond. Sci. Technol. 16, L33 (2001)
- 6 Q. Jiang, H.X. Shi, M. Zhao: J. Chem. Phys. 111, 2176 (1999)
- 7 Z. Wen, M. Zhao, Q. Jiang: J. Phys.: Condens. Matter 12, 8819 (2000)
- 8 Q. Jiang, H.X. Shi, J.C. Li: Thin Solid Films 354, 283 (1999)
- 9 Z. Zhang, M. Zhao, Q. Jiang: Physica B 293, 232 (2001)
- 10 G. Arlt: J. Mater. Sci. 25, 2655 (1990)
- 11 K. Uchino, Y. Sadanaga, T. Hirose: J. Am. Ceram. Soc. 72, 1555 (1989)
- 12 W.L. Zhong, P.L. Zhang, H.C. Chen, F.S. Chen, Y.Y. Song: Ferroelectrics 74, 325 (1987)
- 13 T. Mitsui, I. Tatsuzaki, E. Nakamira: An Introduction to the Physics of Ferroelectrics (Gordon and Breach, New York 1974) p. 151
- 14 K. Ishikawa, K. Yoshikawa, N. Okada: Phys. Rev. B 37, 58 (1988)
- W.L. Zhong, B. Jiang, P.L. Zhang, J.M. Ma, H.M. Chen, Z.H. Yang, L. Li: J. Phys.: Condens. Matter 5, 2619 (1993)
- 16 J.K. Vij, A.M. Varaprasad: Ferroelectrics 38, 865 (1981)
- 17 K.H. Hellwege (Ed.): Landolt-Börnstein New Series on Crystals and Solid State Physics, Group III, Vol. 16/a (Springer-Verlag, Berlin 1981) p. 77; pp. 122–125
- 18 S. Stemmer, S.K. Streiffer, F. Ernst, M. Ruhle: Philos. Mag. A 71, 713 (1995)