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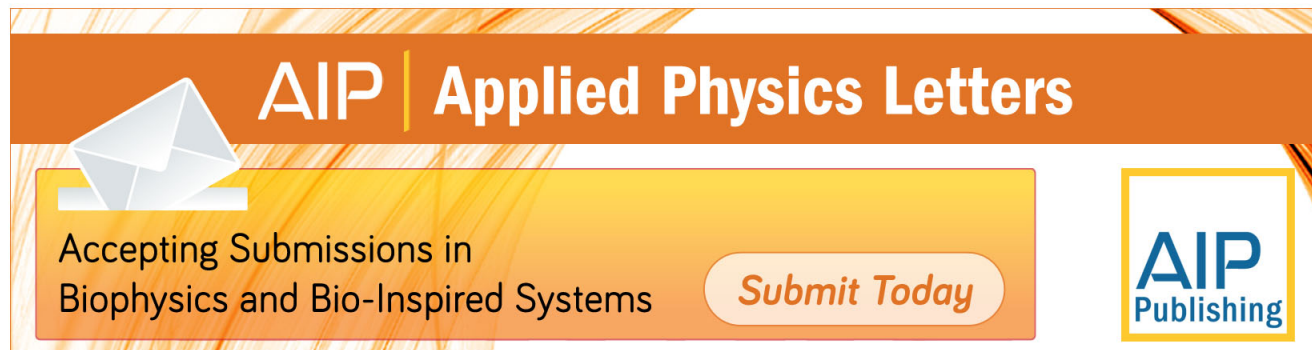
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# High Curie temperature piezocrystals in the $\text{BiScO}_3\text{-PbTiO}_3$ perovskite system

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Perovskite ferroelectric single crystals in the  $(1-x)\text{BiScO}_3\text{-}x\text{PbTiO}_3$  (BSPT) system were grown using a flux method. The Curie temperature of rhombohedral BSPT(43/57) crystals was found to be about 404 °C, with a rhombohedral–tetragonal phase transition temperature ( $T_{r-t}$ ) around 350 °C. The coercive field ( $E_c$ ) was found to be 13.7 kV/cm for  $\langle 001 \rangle$ -oriented crystals, while the coupling factor  $k_{33}$  was determined to be  $\sim 90\%$  at room temperature and was nearly temperature independent until 330 °C. High piezoelectric properties together with a high Curie temperature and coercive field make rhombohedral BSPT crystals promising candidates for the next generation of high-performance, high-temperature actuators and transducers. © 2003 American Institute of Physics. [DOI: 10.1063/1.1619207]

For piezoelectric polycrystalline materials, compositions close to a morphotropic phase boundary (MPB) between rhombohedral (R) and tetragonal (T) phases in the solid-solution  $\text{PbZrO}_3\text{-PbTiO}_3$  (PZT) are widely utilized in transducer and actuator applications.<sup>1,2</sup> Since the discovery of PZTs in the 1950s, extensive research has been carried out for the search of new MBP systems. Recently,  $\text{Pb}(\text{Zn}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{-PbTiO}_3$  (PZNT) and  $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})\text{O}_3\text{-PbTiO}_3$  (PMNT) systems have attracted interest in the single-crystal form<sup>3–7</sup> because of their ultrahigh piezoelectric and electromechanical coupling factors in the  $\langle 001 \rangle$  direction, which can reach  $d_{33s} > 2000$  pC/N and  $k_{33} \sim 94\%$ , respectively. This enhanced piezoelectric activity, however, comes with the sacrifice of the temperature stability of the properties, being limited to their relatively low Curie temperatures ( $T_C \sim 140\text{--}170$  °C), and further restricted by their lower phase transition temperatures ( $T_{r-t} \sim 60\text{--}120$  °C), the consequence of a strong curvature in their MPBs.

In most applications, especially in the automotive and aerospace industries, the need for actuation and sensing over a broad temperature range is essential.<sup>8</sup> Systems with enhanced piezoelectric activity and higher Curie temperature have redirected effort on single crystals in relaxor-PT systems, including  $\text{Pb}(\text{Sc}_{1/2}\text{Nb}_{1/2})\text{O}_3\text{-PbTiO}_3$ ,<sup>9</sup>  $\text{Pb}(\text{In}_{1/2}\text{Nb}_{1/2})\text{O}_3\text{-PbTiO}_3$ ,<sup>10</sup> and  $\text{Pb}(\text{Yb}_{1/2}\text{Nb}_{1/2})\text{O}_3\text{-PbTiO}_3$  (PYNT).<sup>11–13</sup> In particular, PYNT offer the highest  $T_C$  among all the relaxor-PT systems studied to date with a  $T_C \sim 350$  °C. However, as found in PMNT and PZNT single crystals, a strongly curved MPB limits the use of the materials at the  $T_{r-t} \sim 160\text{--}170$  °C.<sup>13</sup> Recently, high-Curie-temperature ferroelectric materials based on  $(1-x)\text{BiMeO}_3\text{-}x\text{PbTiO}_3$  solid solutions (where  $\text{Me}^{3+} = \text{Sc}, \text{In}, \text{Yb}$ , etc.) have been identified.<sup>14</sup> Specifically,  $(1-x)\text{BiScO}_3\text{-}x\text{PbTiO}_3$  (BSPT) exhibits a MPB at  $x = 0.64$  with a  $T_C \sim 450$  °C. Piezoelectric coefficients on the order of 500 pC/N have been demonstrated in polycrystalline ceramics comparable to soft PZTs.<sup>15</sup> In this work, the dielectric and

piezoelectric properties of BSPT single crystals in the R phase close to the MPB were investigated.

BSPT(43/57) single crystals were grown using the high temperature solution method in which  $\text{Pb}_3\text{O}_4$  and  $\text{Bi}_2\text{O}_3$  were selected as a self-flux.<sup>16</sup> The as-grown crystals were blue-green in color and rectangular in shape, being 3–8 mm in size. The samples for measurement were oriented along the crystallographic direction  $\langle 001 \rangle$  and sputtered with gold. The samples were poled at 120 °C with the application of a 20–30 kV/cm electric field for 10 mins. The room-temperature dielectric permittivity ( $\epsilon_r$ ) of  $\langle 001 \rangle$ -oriented BSPT(43/57) crystals was found to be  $\sim 3000$ , with a dielectric loss  $\sim 4\%$  after the poling process. Dielectric permittivity and loss as a function of temperature were performed from room temperature to 500 °C using a multifrequency LCR meter (HP4284A) and are shown in Fig. 1. A dielectric maximum was observed at 404 °C, which was assumed to be the Curie temperature, corresponding well to polycrystalline ceramic data.<sup>15</sup> A second dielectric anomaly in Fig. 1 was observed at 349 °C, being the result of a R–T ferroelectric phase transition. It should be noted that the dielectric permit-

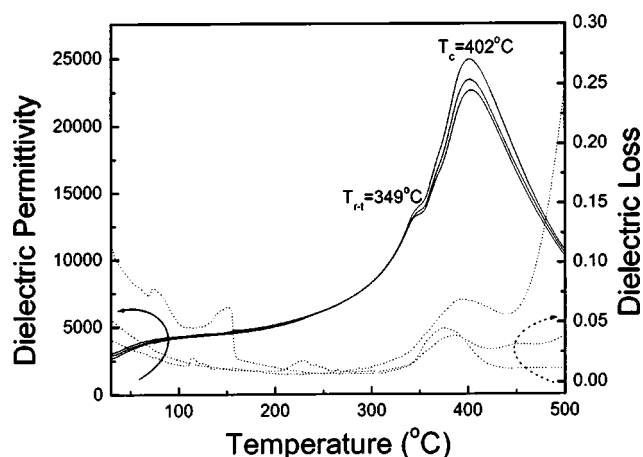


FIG. 1. Dielectric permittivity and loss as a function of temperature for  $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals.

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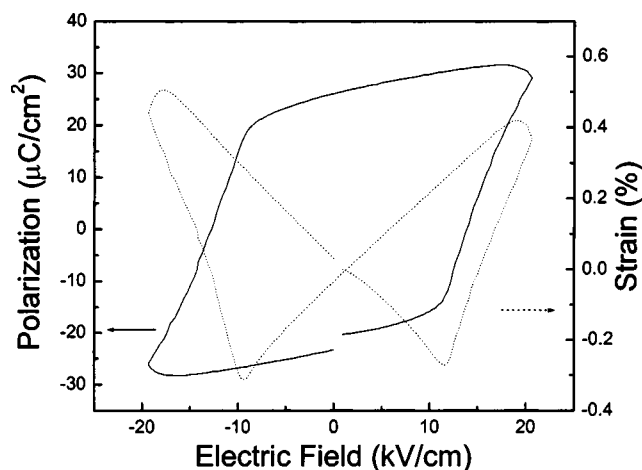


FIG. 2. Bipolar  $P$ - $E$  loops and  $S$ - $E$  loops for  $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals.

tivity is nearly temperature independent in the range of room temperature to  $300^\circ\text{C}$  ( $\partial\epsilon/\partial T \sim 20/^\circ\text{C}$ ), indicating thermal stability associated with a high  $T_C$  piezoelectrics. The dielectric loss was found to decrease to 2% in the temperature range of 150 to  $350^\circ\text{C}$  and reached a peak at the Curie temperature, after which it increased rapidly owing to conductive losses.

Typical hysteresis and strain-field loops determined using a modified Sawyer-Tower circuit driven by a lock-in amplifier (Stanford Research System, Model SR830) at 1 Hz and 20 kV/cm electric fields for  $\langle 001 \rangle$ -oriented BSPT(43/57) crystals were measured, and are presented in Fig. 2. The coercive field ( $E_c$ ) was found to be  $\sim 13.7$  kV/cm with a remnant polarization of  $23 \mu\text{C}/\text{cm}^2$ . The coercive field is significantly higher than that for PZNT and PMNT systems, being on the order of 2–3 kV/cm.<sup>12</sup> The unipolar strain behavior of  $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals exhibited low hysteresis, with strains reaching 0.12% at 10 kV/cm, as shown in Fig. 3. The piezoelectric coefficient  $d_{33}$  calculated directly from strain-field curve was  $\sim 1200$  pC/N, similar to that measured by using a Berlincourt  $d_{33}$  meter. A maximum strain value of  $\sim 1.1\%$  was reached with an electric field of 30 kV/cm, as the crystal transformed from the R to T state with large hysteresis.

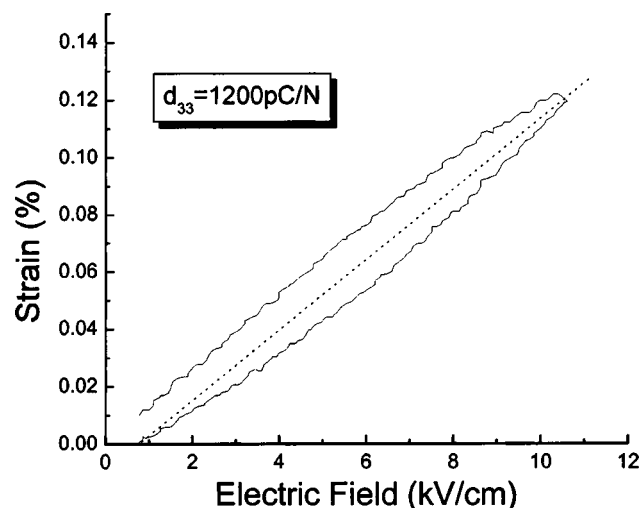


FIG. 3. Unipolar  $S$ - $E$  loop for  $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals.

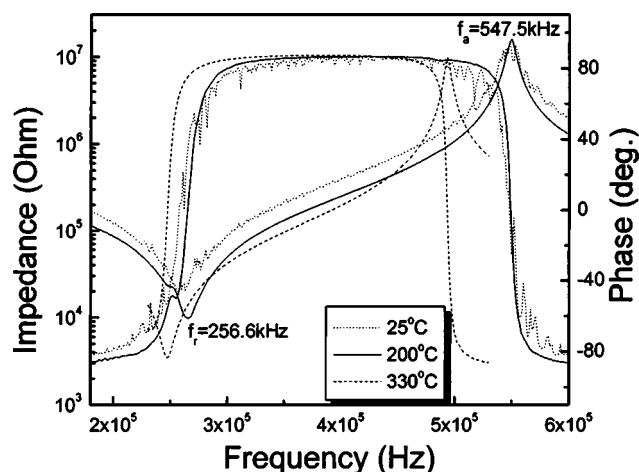


FIG. 4. Resonance and antiresonance frequency characteristics of impedance and phase for the  $\langle 001 \rangle$ -oriented BSPT(43/57) single crystals with different temperatures.

The resonance ( $f_r$ ) and antiresonance ( $f_a$ ) frequencies for a  $\langle 001 \rangle$ -oriented BSPT(43/57) rod (size:  $0.6 \times 0.6 \times 3.05 \text{ mm}^3$ ) measured using an impedance analyzer (HP4194A) were used to determine the electromechanical coupling factor. As shown in Fig. 4, the  $f_r$  and  $f_a$  were found to be 256.6 and 547.5 kHz at room temperature, respectively, and the longitudinal electromechanical coupling factor  $k_{33}$  was calculated to be 90%. The calculated piezoelectric coefficient  $d_{33}$  of 1150 pC/N was in good agreement with the values obtained from the unipolar strain measurement. As a function of temperature in the range of 25 to  $320^\circ\text{C}$ , both  $f_r$  and  $f_a$  shifted a little bit with  $k_{33}$  decreasing to 89%. Above  $330^\circ\text{C}$ ,  $f_a$  shifted lower and the coupling factor decreased to 88.5%, decreasing further to only 83% when the temperature reached  $340^\circ\text{C}$ , as shown in Fig. 5, corresponding to the value of coupling factor for BSPT tetragonal crystal.<sup>16</sup> The same tendencies for the piezoelectric coefficients were also observed and are shown in Fig. 5. This sudden drop above  $340^\circ\text{C}$  in the coupling factors and piezoelectric coefficients were the result of the R-T phase transformation as evident in the dielectric behavior.

The various dielectric, piezoelectric, and electromechanical property characterizations of BSPT(43/57) crystals

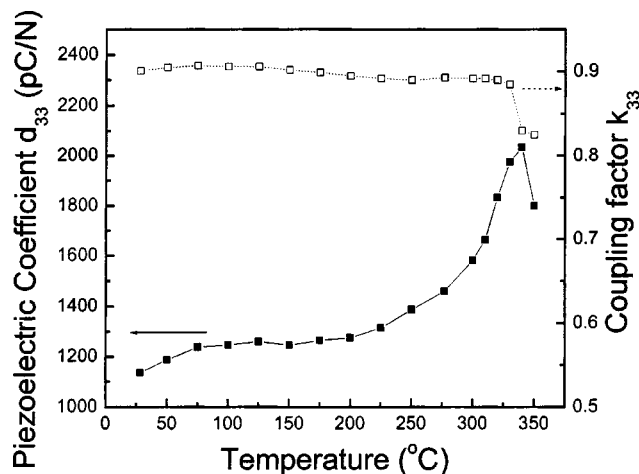


FIG. 5. Longitudinal electromechanical coupling factors  $k_{33}$  and piezoelectric coefficients  $d_{33}$  as a function of temperature.

TABLE I. Dielectric, piezoelectric, and electromechanical properties of the (001)-oriented BSPT(43/57) single crystals compared with (001)-oriented relaxor-PT single crystals and BSPT ceramics.

	$T_C$ (°C)	$T_{r-t}$ (°C)	$K_{33}^T$	Loss	$P_r$ ( $\mu\text{C}/\text{cm}^2$ )	$E_c$ (kV/cm)	$d_{33}$ (pC/N)	$k_{33}$
BSPT57 crystal <sup>a</sup>	402	349	3000	4%	23	13.7	1150	90%
BSPT64 ceramic <sup>b</sup>	450	/	2010	5%	32	20	460	/
PYNT40 crystal <sup>c</sup>	270	168	2700	2%	26.5	10	1200	88%
PMNT30 crystal	138	98	5000	0.5%	26	2.3	1600	89%
PZNT4.5 crystal	155	120	5000	0.5%	27	3	1800	90%
PZNT8 crystal	170	90	6000	0.5%	30	4	2500	93%

<sup>a</sup>This work.<sup>b</sup>See Ref. 15.<sup>c</sup>See Ref. 17.

are presented in Table I. For comparison, values are also given for various relaxor-PT crystals and polycrystalline BSPT ceramics.

In summary, rhombohedral BSPT(43/57) single crystals possess a high Curie temperature  $>400^\circ\text{C}$  and phase transition temperature  $>340^\circ\text{C}$ , with coercive fields greater than 13 kV/cm. Together with high piezoelectric coefficient  $d_{33} \sim 1200$  pC/N, coupling  $k_{33} > 90\%$ , and temperature-independent properties, BSPT single crystals are attractive candidates for the next-generation high-temperature, high-performance actuators and transducers.

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<sup>1</sup>B. Jaffe, J. Res. Natl. Bur. Stand. **55**, 239 (1955).<sup>2</sup>R. E. Newnham, *Ferroelectric Ceramics*, edited by N. Setter (Birkh Suser, Basel, 1992) p. 363.<sup>3</sup>S.-E. Park and T. R. Shrout, J. Appl. Phys. **82**, 1804 (1997).<sup>4</sup>D. Viehland, J. Powers, and L. E. Cross, Appl. Phys. Lett. **78**, 3508 (2001).<sup>5</sup>H. S. Luo, G. S. Xu, and Z. W. Yin, Jpn. J. Appl. Phys. **39**, 5581 (2000).<sup>6</sup>S. J. Zhang, L. Lebrun, S. F. Liu, S. Rhee, C. A. Randall, and T. R. Shrout, Jpn. J. Appl. Phys. **41**, L1099 (2002).<sup>7</sup>S. J. Zhang, L. Lebrun, D. Y. Jeong, C. A. Randall, Q. M. Zhang, and T. R. Shrout, J. Appl. Phys. **93**, 9257 (2003).<sup>8</sup>T. R. Shrout, R. Eitel, and C. A. Randall, *Piezoelectric Materials in Devices*, edited by N. Setter (EPFL Swiss Federal Institute of Technology, Lausanne, Switzerland, 2002), p. 413.<sup>9</sup>Y. H. Bing and Z. G. Ye, J. Cryst. Growth **250**, 118 (2003).<sup>10</sup>N. Yasuda, H. Ohwa, M. Kume, and Y. Yamashita, J. Cryst. Growth **229**, 299 (2001).<sup>11</sup>N. Yasuda, H. Ohwa, M. Kume, Y. Hosono, Y. Yamashita, S. Ishino, H. Terauchi, M. Iwata, and Y. Ishibashi, Jpn. J. Appl. Phys. **40**, 5664 (2001).<sup>12</sup>S. J. Zhang, S. Rhee, C. A. Randall, and T. R. Shrout, Jpn. J. Appl. Phys. **41**, 722 (2002).<sup>13</sup>S. J. Zhang, L. Lebrun, S. Rhee, C. A. Randall, and T. R. Shrout, Appl. Phys. Lett. **81**, 892 (2002).<sup>14</sup>R. Eitel, C. A. Randall, T. R. Shrout, P. Rehrig, W. Hackenberger, and S.-E. Park, Jpn. J. Appl. Phys. **40**, 5999 (2001).<sup>15</sup>R. Eitel, C. A. Randall, T. R. Shrout, and S. E. Park, Jpn. J. Appl. Phys. **41**, 2099 (2002).<sup>16</sup>S. J. Zhang, L. Lebrun, S. Rhee, R. E. Eitel, C. A. Randall, and T. R. Shrout, J. Cryst. Growth **236**, 210 (2002).<sup>17</sup>S. J. Zhang, L. Lebrun, S. Rhee, C. A. Randall, and T. R. Shrout, *Proceedings of the 13th IEEE International Symposium on Applications of Ferroelectrics*, edited by G. White and T. Tsurumi (IEEE, Piscataway, NJ, 2002), p. 455.