



Modeling of multifunctional donor-bridge-acceptor 4,6-di(thiophen-2-yl)pyrimidine derivatives: A first principles study



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ABSTRACT

We have modeled multifunctional compounds by pi-elongation and push–pull strategy from the 4,6-di(thiophen-2-yl)pyrimidine. The ground state geometries have been optimized by density functional theory while excited state geometries were optimized by time dependent density functional theory (TDDFT). Structure–property relationship, electronic, optical and charge transfer properties (ionization potential, electron affinity and reorganization energies) were computed and discussed. By TDDFT absorption and emission have been calculated. The computed parameters were compared with available experimental data. The long-range corrected functional (LC-BLYP) is overestimating the highest occupied and lowest unoccupied molecular orbital energies, energy gaps while underestimating the absorption and fluorescence wavelengths. The B3LYP is good to reproduce the experimental data. The intra-molecular charge transfer has been observed from highest occupied molecular orbitals to lowest unoccupied molecular orbitals. The strong electron withdrawing and electron donor groups efficiently reduce the energy gaps. The decrease injection barrier and smaller reorganization energies are revealing that our designed derivatives would be efficient hole as well as electron transfer materials. These derivatives would be good light emitters e.g. blue, green, orange, red and near IR. The predicted values showed that the designed derivatives would be efficient for the organic field effect transistors, photovoltaics and light emitters.

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1. Introduction

Organic electronics are emergent rapidly in important technological fields [1–4]. Conjugated materials are being used in batteries [5–9], electroluminescent devices [10], field-effect transistors [11], and photovoltaics [12], etc. The polymers or small molecules are attracting attention due to the low-cost, environment friendly, ease to fabricate and their flexibility than traditional materials, like silicon. Such properties make organic materials ultimate for applications in mobile electronic devices as well. The organic electronic materials have attracted passionate and unremitting interest in recent years. To enhance the charge carrier mobilities and/or the quantum efficiency of organic electronics is under consideration. Oligothiophenes are well-characterized conjugated systems which have been investigated previously [13–16]. These semiconductors are important contenders which have been used in molecular electronics, as light emitting diodes and field-effect transistors [17,18]. The band gap is a way to control the electrical properties, which

are strongly governed by the intra-molecular delocalization of π -electrons along the conjugation chain [19]. In optoelectronic devices pi-conjugated oligomers and polymers based on heteropentacenes containing pyrrole, thiophene, and silole have been widely used [20–22]. The OFETs can now compete with other thin film transistors such as amorphous silicon devices [23].

The first organic light-emitting field-effect transistor (OLFET) has been reported in 2003 [24] which is a new class of electro-optical devices. The OLFETs could provide a novel architecture to address open questions concerning charge-carrier recombination and light emission in organic materials [25]. The split-gate LFETs demonstrate a new strategy for ambipolar LFETs to achieve high brightness and efficiency simultaneously [26]. Two active layers of pentacene and *N,N'*-ditridecylperylene-3,4,9,10-tetra carboxylic diimide as well as a protecting layer of 2,5-bis(4-biphenyl) thiophene were studied as OLFETs previously [27]. The 4,6-di(thiophen-2-yl)pyrimidine (**1**) has alternate pi-rich and pi-poor units, i.e., thiophene and pyrimidine [28]. We have designed new derivatives of **1** with thiophene, pyrimidine and oligocene units with the aim to enhance optical and charge transfer properties.

Different computational studies are effective tools to design of efficient functional materials (like organic electronic materials),

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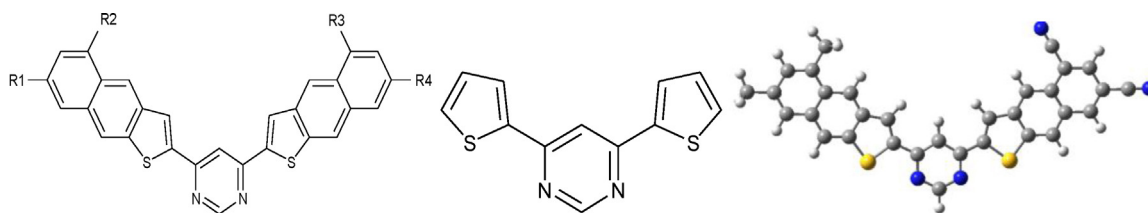
**1** (centre)**2** = R1, R2 = CH₃; R3, R4 = CN**3** = R1, R2 = CH₃; R3, R4 = COOH**4** = R1, R2 = CH₃; R3, R4 = F**5** = R1, R2 = CH₃; R3, R4 = NO₂**6** = R1, R2 = OCH₃; R3, R4 = CN**7** = R1, R2 = OCH₃; R3, R4 = COOH**8** = R1, R2 = OCH₃; R3, R4 = F**9** = R1, R2 = OCH₃; R3, R4 = NO₂**10** = R1, R2 = OH; R3, R4 = CN**11** = R1, R2 = OH; R3, R4 = COOH**12** = R1, R2 = OH; R3, R4 = F**13** = R1, R2 = OH; R3, R4 = NO₂

Fig. 1. The optimized structures of **1** (center) and its derivatives investigated in the presented study (the structure at right is an example of derivative, i.e., **2**).

to predict their properties and to explain the available experimental data. New derivatives have been designed by push-pull strategy and elongating the bridge with the aim to reduce the HOMO–LUMO energy gap, improved intra-molecular charge transfer which would tailor the electronic, optical and charge transfer properties, see Fig. 1. The effect of electron activating groups, deactivating groups and bridge has been studied on the geometries, dipole moments, electronic, spectroscopic (absorption and fluorescence), and charge transfer properties (ionization potentials, electron affinities, reorganization energies) and discussed thoroughly. The structure–property relationship has been discussed. The paper is structured as follows: Section 2 presents an outline of the density functional theory (DFT) and time dependent density functional theory (TDDFT) methodology used, including the rationale for choosing the hybrid functional and the basis set; Section 3 gives the frontier molecular orbitals, electronic, optical and charge transfer properties with detailed comparisons with experimental and theoretical studies; in Section 4 the major conclusions of the present investigation are presented.

2. Computational details

The line shape of the lowest gas-phase ultraviolet photoelectron spectroscopy (UPS) band is directly related to the

geometry relaxation energy thus to the polaron binding energy [29]. Among the standard DFT [30] functionals, B3LYP provides the best description of the geometry modifications upon ionization [31]. The ground state (*S*₀) geometries have been optimized using density functional theory (DFT). The Becke's three parameter gradient-corrected exchange potential and the Lee–Yang–Parr gradient-corrected correlation potential (B3LYP) [32–34], and 6-31G** basis set [35] has been used for all the calculations. The frequencies have been computed at the same level of theory and no imaginary frequency has been observed. The absorption spectra have been computed by using time dependent density functional theory (TDDFT) which has been proved an efficient approach [36,37]. The excited state (*S*₁) geometries have been optimized at TDDFT [38–44] using B3LYP/6-31G** level of theory. The emission spectra have been computed by using TD-B3LYP [45] and 6-31G** basis set [35].

The efficiency of charge transport within the organic layer(s) plays a key role. The high-charge mobilities favor recombination processes in the bulk where charges can be confined further by means of organic–organic interfaces [46]. The charge transfer rate can be described by Marcus theory via the following equation [47]:

$$W = \frac{V^2}{h(\pi/\lambda k_B T)^{1/2}} \exp\left(-\frac{\lambda}{4k_B T}\right) \quad (1)$$

There are two major parameters that determine self-exchange electron-transfer rates and ultimately charge mobility: (i) the electronic coupling V (transfer integral) between adjacent molecules, which needs to be maximized and (ii) the reorganization energy λ , which needs to be small for significant transport. The reorganization energy term describes the strength of the electron–phonon (vibration) and can be reliably estimated as twice the relaxation energy of a polaron localized over a single unit.

The reorganization energy is further divided into two parts: $\lambda_{\text{rel}}^{(1)}$ and $\lambda_{\text{rel}}^{(2)}$, where $\lambda_{\text{rel}}^{(1)}$ corresponds to the geometry relaxation energy of one molecule from neutral to charged state, and $\lambda_{\text{rel}}^{(2)}$ corresponds to the geometry relaxation energy from charged to neutral state [48]:

$$\lambda = \lambda_{\text{rel}}^{(1)} + \lambda_{\text{rel}}^{(2)} \quad (2)$$

In the evaluation of λ , the two terms were computed directly from the adiabatic potential energy surfaces [49,50]:

$$\lambda = \lambda_{\text{rel}}^{(1)} + \lambda_{\text{rel}}^{(2)} = [E^1(Y^+) - E^0(Y^+)] + [E^1(Y) - E^0(Y)] \quad (3)$$

Here, $E^0(Y)$, $E^0(Y^+)$ are the ground-state energies of the neutral and charged states, $E^1(Y)$ is the energy of the neutral molecule at the optimized charged geometry and $E^1(Y^+)$ is the energy of the charged state at the geometry of the optimized neutral molecule. It should be noted that the polarization effects from the surrounding molecules, as well as the charge reorientation, have been neglected to minimize the complications involved in the theoretical calculations [51,52]. The adiabatic ionization potential (IPa) and vertical ionization potential (IPv) have been calculated as:

$$\text{IPa} = E^0(Y) + -E^0(Y) \quad \text{and} \quad \text{IPv} = E^1(Y)^+ - E^0(Y).$$

The ground state energies of the neutral and cation states represented by $E^0(Y)$ and $E^0(Y)^+$ respectively, and $E^1(Y)^+$ is the energy of cation state at the optimized geometry of the neutral molecule. The adiabatic/vertical electron affinity (EAa)/(EAv) of all molecules has been calculated as under:

$$\text{EAa} = E^0(Y) - E^0(Y)^- \quad \text{and} \quad \text{EAv} = E^0(Y)^- - E^1(Y)^-$$

Here, $E^0(Y)$ corresponds to the ground state energies of the neutral and $E^0(Y)^-$ is the energy of anion states. The term $E^1(Y)^-$ represents the energy of anion state at the optimized geometry of the neutral molecule. All the calculations have been performed by using Gaussian 09 package [53].

Generally, it is esteemed that B3LYP is not to good reproduce the experimental data, i.e., electronic properties and excitation energies, etc. The excitation energies of coumarin-based dyes were reproduced by Stein et al. by using a range-separated hybrid functional [54]. In push–pull oligothiophene biomarkers the absorption and fluorescence wavelengths were studied by long-range-corrected (LC) method [55]. The LC-BLYP has been used to investigate the excited-state properties in a series of coumarin dyes [56]. The range-separation technique is based on a more physical model of the exchange potential. The B3LYP hybrid functional underestimates the vertical excitation energies, especially for larger molecules. It has been already reported that the choice of the range separation parameter is strongly system dependent [57–60]. Moreover, B3LYP/6-31G(d,p) level is adequate which has been applied to computed the different properties and successfully reproduce the experimental data previously [61–65,34,66].

To validate B3LYP for our studied systems, we have also optimized the geometries at ground and excited states by using long-range corrected functional (LC-BLYP). Then we have also computed the absorption and emission wavelengths. Benchmark computational study has been carried out on parent molecule. In first step, we have compared the HOMO energy, LUMO energy and

HOMO–LUMO energy gap absorption, and fluorescence of parent molecule with available experimental data at B3LYP/6-31G** and TD-B3LYP/6-31G**//B3LYP/6-31G** level of theories. In next step all above mentioned properties were also computed and compared at TD-LC-BLYP/6-31G**//B3LYP/6-31G** level of theory. Then we have also computed and compared the data at LC-BLYP/6-31G** and TD-LC-BLYP/6-31G**//LC-BLYP/6-31G** level of theory. It can be seen from Sections 3.1 and 3.2 that B3LYP is good choice to reproduce the experimental data.

3. Results and discussion

3.1. Electronic structure of the ground and first excited states

The distribution patterns of the highest occupied molecular orbitals (HOMOs) and lowest unoccupied molecular orbitals (LUMOs) for the ground states have been shown in Fig. 2. In 1, the HOMO and LUMO are distributed throughout the backbone. In designed derivatives HOMOs are delocalized toward the electron activating groups, i.e., left side of the systems. In the formation of HOMOs pyrimidine and thiophene rings are also taking part. The LUMOs are localized toward the electron deactivating groups, i.e., pyrimidine, thiophene and benzene rings at right side. The HOMOs and LUMOs are also spread over the electron activating as well as electron deactivating groups, respectively. In the formation of HOMOs strong deactivate group has more participation than the LUMO. In ground state comprehensible intra-molecular charge transfer has been observed from donor (push) to acceptor (pull) moieties. Fig. 3 illustrates the distribution patterns of the HOMOs and LUMOs at excited states. The HOMOs are delocalized at the left side while LUMOs are localized toward the right side. The pyrimidine and thiophene rings of the right side are also taking part in the formation of HOMOs. The LUMOs are localized toward the electron deactivating groups, i.e., pyrimidine, thiophene and benzene rings at right side. Generally, the HOMOs and LUMOs are distributed on the electron activating and electron deactivating groups. The distribution pattern of the HOMOs and LUMOs at the excited states is alike at the ground states. Moreover, we have noticed that at ground and excited states, the LUMOs are also localized on that moiety which is toward the acceptor group, i.e., strong deactivating group. The thiophene and benzene toward the activating group have no participation in the formation of LUMOs.

The computed HOMO energy, LUMO energy and HOMO–LUMO energy gap of parent molecule (1) at ground state by using B3LYP/6-31G** level of theory are –6.19, –1.94 and 4.25 eV, respectively. The calculated HOMO energy, LUMO energy and HOMO–LUMO energy gap are in good agreement with the experimental data, i.e., –5.30, –1.90, and 3.40 eV [28]. While the computed HOMO energy, LUMO energy and HOMO–LUMO energy gap at LC-BLYP/6-31G** level of theory are –9.05, 0.34 and 9.39 eV revealing that LC-BLYP/6-31G** level of theory would be not a good choice for our studied systems. In the next step, we have computed the HOMO energies, LUMO energies and HOMO–LUMO energy gaps of new designed derivatives at the B3LYP/6-31G** level of theory.

The HOMO energies of new designed derivatives are higher while LUMO energies are lower than the parent molecule at both ground and excited states, respectively. The HOMO–LUMO energy gap in new designed systems is smaller than the parent molecule. By introducing the stronger activating groups higher the HOMO energies in the studied derivatives. The trend to higher the HOMO energies on the base of strength of activating group is as $\text{OCH}_3 > \text{OH} > \text{CH}_3$ both at ground and excited states. On other hand, the trend to lower the LUMO energies on the base of strength of activating group is as $\text{OCH}_3 < \text{OH} < \text{CH}_3$ both at ground and excited states. The trend to higher the HOMO energies on the base of

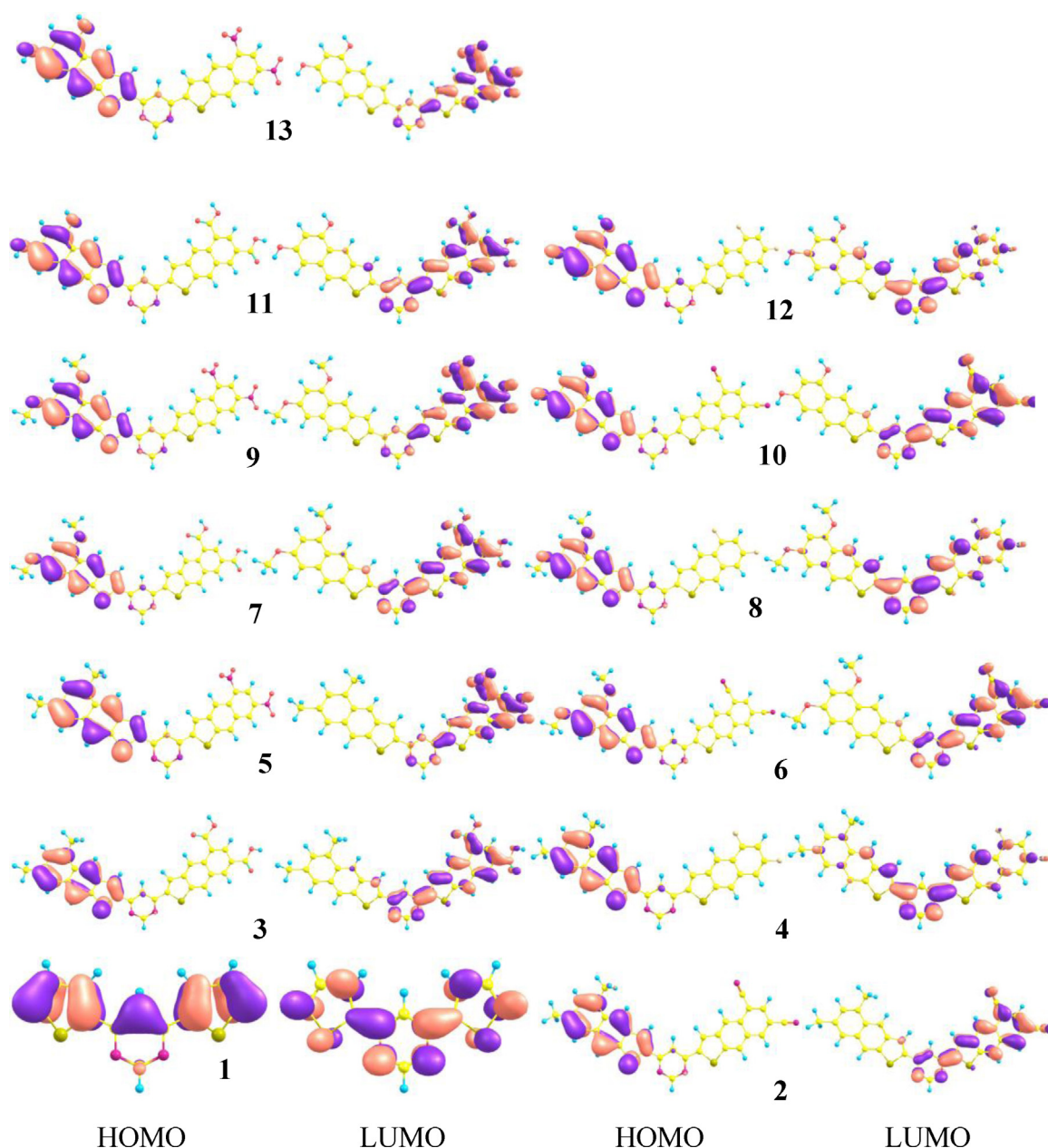


Fig. 2. Distribution patterns of the HOMO and LUMO of the ground state.

strength of deactivating group is as $F > \text{COOH} > \text{CN} > \text{NO}_2$ while tendency to lower the LUMO energies is as $F > \text{COOH} > \text{CN} > \text{NO}_2$ both at ground and excited states. The smallest HOMO–LUMO energy gaps have been observed for those systems which have NO_2 or CN substituted groups. Such derivatives are effected more which have the strong activating and deactivating groups resulting smaller HOMO–LUMO energy gaps, i.e., **5** (2.26 eV) > **13** (2.12 eV) > **9** (2.06 eV).

In Table 1, we listed the calculated energies for the HOMOs (E_{HOMO}), LUMOs (E_{LUMO}) and HOMO–LUMO energy gaps (E_{gap}) for the **1** and its derivatives at ground and first excited states. We found that the computed E_{HOMO} , E_{LUMO} and E_{gap} are in good agreement with the experimental data [28]. The trend of HOMO energies (E_{HOMO}) is as follows: **8** > **7** > **12** > **11** > **6** > **9** > **10** > **4** > **13** > **3** > **2** > **5** > **1** while in LUMOs energies (E_{LUMO}) is as: **1** > **8** > **12** > **4** > **7** > **11** > **3** > **6** > **10** > **2** > **9** > **13** > **5**. The trend of E_{gap} in studied compounds is **1** > **4** > **12** > **8** > **3** > **11** > **7** > **2** > **10** > **6** > **5** > **13** > **9**, see Table 1. From

the trend of HOMO–LUMO energy gaps, we found that stronger activating and deactivating groups reduced the energy gaps more efficiently.

The work function of aluminum is -5.1 eV and the LUMO energy level of **1** is -1.94 eV (see Table 1), the electron injection energy is around 3.16 eV ($= -1.94 - (-5.1)$) from the **1** to Al electrode. Thus it is necessary to lower the LUMO level to improve the electron injection capability. From Table 1, it can be found that LUMO energy levels lowered in new designed derivatives compared to the parent molecule which would lower the injection barrier and improve the electron injection. The electron injection barrier decreases in new designed derivatives as follows: **8** (2.81), **12** (2.78), **4** (2.71), **7** (2.52), **11** (2.50), **3** (2.46), **6** (2.18), **10** (2.16), **2** (2.12), **9** (1.84), **13** (1.83), **5** (1.79). The hole injection energy is around 1.09 eV ($= -5.1 - (-6.19)$) from the **1** to Al electrode. Therefore, to improve the hole injection ability it is necessary to higher the HOMO energy level. The hole injection barrier decreased in all the newly designed derivatives as: **8** (0.08), **7** (0.09), **12** (0.23), **11** (0.24), **6** (0.29), **9** (0.31), **10** (0.35), **4**

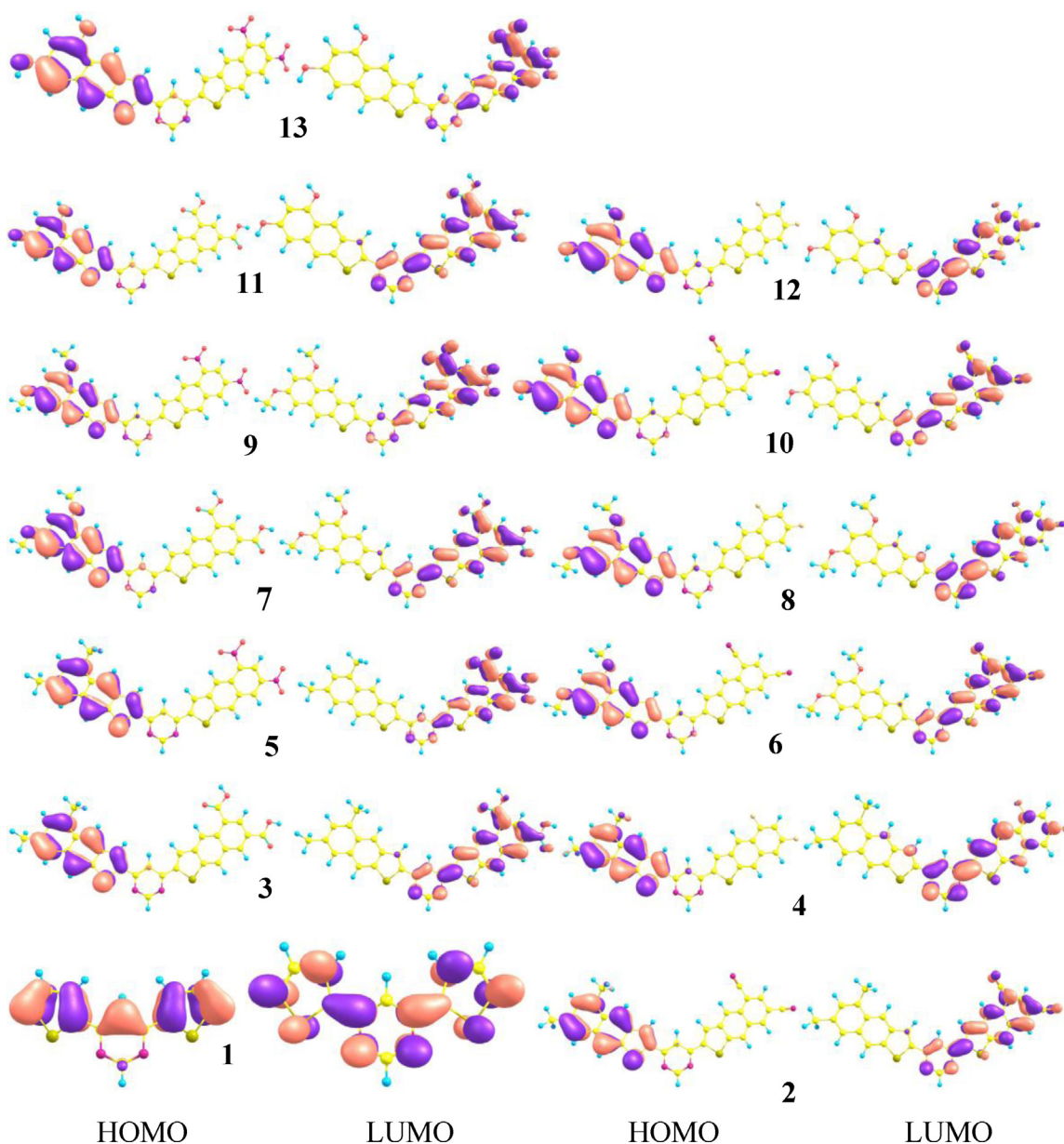


Fig. 3. Distribution patterns of the HOMO and LUMO of the first excited state.

Table 1
The HOMO energies (E_{HOMO}), LUMO energies (E_{LUMO}), and HOMO–LUMO energy gaps (E_{gap}) in eV for ground and first excited states computed at the B3LYP/6-31G** and TD-B3LYP/6-31G** level of theories, respectively.

Complexes	Ground state			First excited state		
	E_{HOMO}	E_{LUMO}	E_{g}	E_{HOMO}	E_{LUMO}	E_{g}
1	−6.19 (−5.30)	−1.94 (−1.90)	4.25 (3.40)	−6.02	−2.10	3.92
2	−5.51	−2.98	2.53	−5.34	−3.25	2.09
3	−5.42	−2.64	2.78	−5.26	−2.96	2.30
4	−5.38	−2.39	2.99	−5.21	−2.64	2.57
5	−5.57	−3.31	2.26	−5.41	−3.62	1.79
6	−5.30	−2.92	2.38	−5.06	−3.16	1.90
7	−5.19	−2.58	2.61	−4.96	−2.88	2.08
8	−5.18	−2.29	2.89	−4.94	−2.55	2.39
9	−5.32	−3.26	2.06	−5.09	−3.55	1.54
10	−5.36	−2.94	2.42	−5.14	−3.19	1.95
11	−5.25	−2.60	2.65	−5.03	−2.91	2.12
12	−5.24	−2.32	2.91	−5.02	−2.58	2.44
13	−5.39	−3.27	2.12	−5.18	−3.57	1.61

Values in parentheses are experimental data from Ref. [28].

Table 2

The dipole moments (Debye) for **1** and its derivatives of the ground and excited states at the B3LYP/6-31G** level of theory, respectively.

1	2	3	4
3.55 (3.77)	6.16 (5.11)	4.87 (4.22)	3.89 (3.89)
5	6	7	8
6.96 (6.37)	7.91 (7.91)	5.95 (5.66)	5.28 (4.87)
9	10	11	12
8.76 (8.58)	7.41 (6.58)	5.20 (5.20)	4.52 (3.99)
13			
8.27 (7.90)			

The values in parentheses are at excited state.

(0.37), **13** (0.38), **3** (0.41), **2** (0.50), **5** (0.56). The decreased injection barrier for hole and electron revealed that new designed derivatives would be better charge transporter than the parent molecule. The LUMO energies of new designed derivatives are low-lying than the parent molecule revealing that new designed materials would be thermodynamically more stable and charge transport cannot be quenched by losing the electron.

3.2. Dipole moment

In Table 2, we have presented the dipole moments (μ) of the ground and first excited states computed at DFT/B3LYP/6-31G** level of theory. Previously we showed that substituted group affects the dipole moment [45]. In this study, we have investigated the effect of electron deactivating groups, electron activating groups and bridge on the dipole moments. By the enhancement of bridge dipole moment generally increases at both ground and excited states compared to parent molecule. Moreover, by substituting the stronger activating and deactivating groups enhance the dipole moment values. We noticed that the effect of weakly activating and deactivating groups on the dipole moment is very little, i.e., **4** and **12** dipole moment is smallest compared to newly designed derivatives.

3.3. Photophysical properties

The computed wavelengths (nm) for absorption and fluorescence along X-axis and oscillator strengths (f) along Y-axis have been illustrated in Fig. 4. The main absorption wavelengths of parent molecule at TD-LC-BLYP/6-31G**//B3LYP/6-31G** level of theory are 272 nm ($f=0.8100$) and 222 nm ($f=0.1674$). Absorption wavelengths of parent molecule at TD-LC-BLYP/6-31G**//LC-BLYP/6-31G** level of theory are 259 nm ($f=0.8019$) and 215 nm ($f=0.1720$). Fluorescence wavelengths of parent molecule at TD-LC-BLYP/6-31G**//LC-BLYP/6-31G** level of theory are 288 nm ($f=0.8356$) and 222 nm ($f=0.1549$). The absorption and emission (fluorescence) of parent molecule have also been computed at TD-B3LYP/6-31G**//B3LYP/6-31G** level of theory. Two absorption peaks have been observed at 280 and 327 nm. The computed absorption wavelength 327 nm is in good agreement with the experimental absorption wavelength, i.e., 329 nm [28]. Similarly, two peaks for the computed fluorescence wavelengths have been observed at 292 and 353 nm. The calculated fluorescence wavelength at 353 nm is also in sound agreement with the experimental data, i.e., 378 nm [28]. These results are revealing that our adopted methodology is reasonable and accurate choice while LC-BLYP is underestimating the absorption and fluorescence wavelengths for parent molecule. Thus we have computed the absorption and emission spectra for new designed derivatives at the TD-B3LYP/6-31G**//B3LYP/6-31G** level of theory.

The **2** showed absorption peaks at 349, 427, and 554 nm. By introducing the COOH (**3**) in place of CN showed prominent absorption wavelengths peaks at 352, 438 and 502 nm. The F substitution

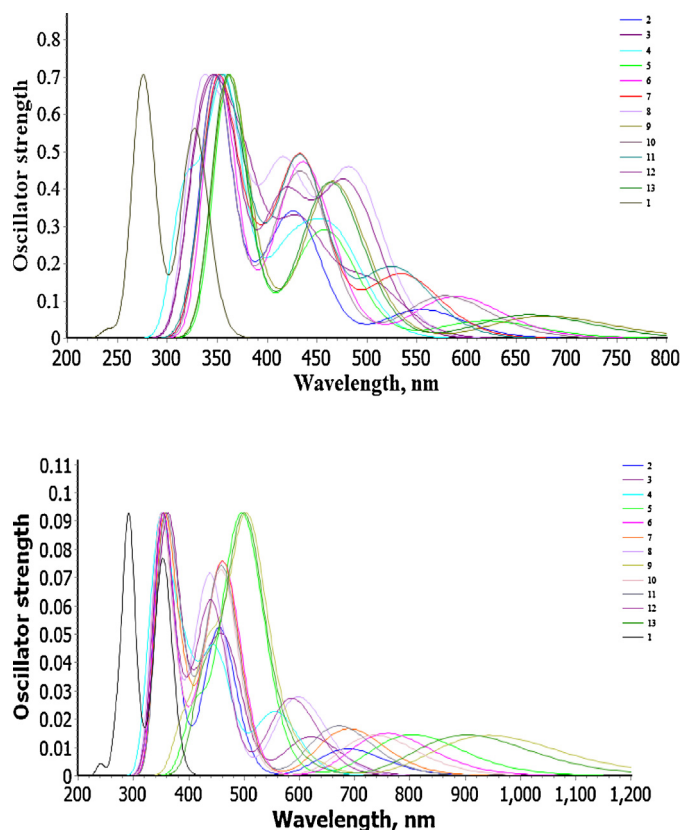


Fig. 4. The absorption (top) and fluorescence (bottom) spectra computed at TD-B3LYP/6-31G** level of theory. Note: The experimental absorption wavelengths (λ_a) = 329 nm; emission wavelength fluorescence (λ_e) = 378 nm from Ref. [28].

(**4**) in place of CN showed the peaks at 359 and 469 nm. The **5** showed peaks at 364, 467 and 619 nm. By substituting the CN and nitro groups the absorption wavelength are being red shifted. The following prominent absorption peaks have been observed for **6** (350, 447 and 588 nm), **7** (354, 438 and 534 nm), **8** (336, 420 and 485 nm) and **9** (365, 470 and 680 nm). The numerous absorption wavelength peaks have been also observed for **10** (345, 444 and 578 nm), **11** (353, 438 and 527 nm), **12** (358, 421 and 481 nm) and **13** (359, 468 and 664 nm).

The **2** showed fluorescence peaks at 350, 460, and 689 nm. By introducing the COOH (**3**) in place of CN showed peaks at 374, 469 and 620 nm. The F substitution (**4**) in place of CN showed the peaks at 352, 451 and 556 nm. The **5** showed peaks at 505 and 804 nm. The following prominent peaks have been observed for **6** (348, 472 and 759 nm), **7** (351, 468 and 689 nm), **8** (343, 447 and 597 nm) and **9** (508 and 944 nm). The numerous absorption wavelength peaks have been also observed for **10** (346, 466 and 737 nm), **11** (352, 467 and 671 nm), **12** (356, 448 and 586 nm) and **13** (431, 505 and 902 nm). The significant stoke shifts from absorption to fluorescence have been observed in the studied compounds. Multiple emission peaks in whole of the visible region and near IR for some compounds revealed that these would be good emitter materials. The new designed derivatives might be better emitters in the following regions; **2** (blue and red), **3** (blue and orange), **4** (blue and green), **5** (green and near IR), **6** (blue and red), **7** (blue and red), **8** (blue and orange), **9** (green and near IR), **10** (blue and red), **11** (blue and red), **12** (blue and orange) and **13** (green and near IR).

3.4. Ionization potential and electron affinity

The ionization potential and electron affinity are good descriptors to shed some light on the charge transfer behavior. The

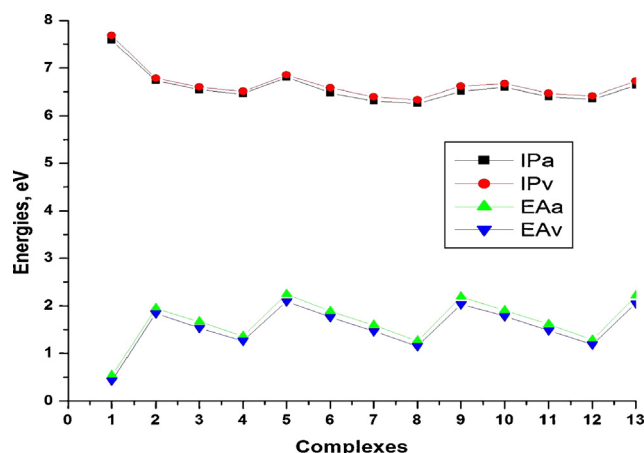


Fig. 5. The vertical and adiabatic ionization potentials (IPv/IPa), vertical and adiabatic electronic affinities (EAa/EAv) (in eV) at the B3LYP/6-31G** level of theory.

ionization potential and electron affinity are distinct properties that can be calculated by DFT to estimate the energy barrier for the injection of holes and electrons. The lower ionization potential revealed that material would be better as hole transporter [67] while higher electron affinity leads to be a better electron transport. We have tabulated the adiabatic ionization potential (IPa), vertical ionization potential (IPv), adiabatic electronic affinity (EAa), and vertical electronic affinity (EAa) in Fig. 5. The IPa/IPv of **2**, **3**, **4** and **5** decreases 0.85/0.90, 1.04/1.08, 1.12/1.17 and 0.78/0.836 eV compared to **1**, respectively. The OCH₃ in place of methyl group further lowers the IPa/IPv, i.e., **6**, **7**, **8** and **9** decreases 1.10/1.10, 1.27/1.28, 1.32/1.34 and 1.07/1.06 eV compared to **1**, respectively resulting hole electron injection would increase. The IPa/IPv of **10**, **11**, **12** and **13** also decreases than that of **1**, i.e., 0.99/1.01, 1.18/1.21, 1.23/1.26 and 0.95/0.96 eV, respectively. The hole injection would increase in the sequence CH₃ < OH < OCH₃. The EAa/EAv of **2**, **3**, **4** and **5** increases 1.40/1.41, 1.12/1.10, 0.82/0.83 and 1.70/1.66 eV, respectively revealing decreased electron injection barrier compared to the parent molecule. In new designed derivatives less energies are required to create the hole which are smaller than the parent molecule illuminating enhanced hole injection. The EAa/EAv of **6**, **7**, **8** and **9** increases 1.30/1.33, 1.05/1.03, 0.72/0.72 and 1.65/1.60 eV, respectively than that of parent molecule. Moreover, the EAa/EAv of **10**, **11**, **12** and **13** increases 1.36/1.35, 1.07/1.05, 0.74/0.75 and 1.67/1.62 eV, respectively than **1**, see Fig. 5. We found that the electron injection decreases by introducing the stronger electron activating groups in the following tendency: CH₃ > OH > OCH₃ while electron injection increases by introducing the stronger electron deactivating groups in the following tendency: F < COOH < CN < NO₂. Thus it is expected that the hole as well as electron injection ability of all the new designed derivatives would be improved than **1**.

3.5. Reorganization energies

The reorganization energy (λ) includes the molecular geometry modifications when an electron is added or removed from a molecule (inner reorganization) as well as the modifications in the surrounding medium due to polarization effects (outer reorganization). Here, we focus inner reorganization energy, which reflects the geometric changes in the molecules when going from the neutral to the ionized state and vice versa. It is the sum of two relaxation energy terms [32]: (1) the difference between the energies of the neutral molecule in its equilibrium geometry and in the relaxed geometry characteristic of the ion and (2) the difference between

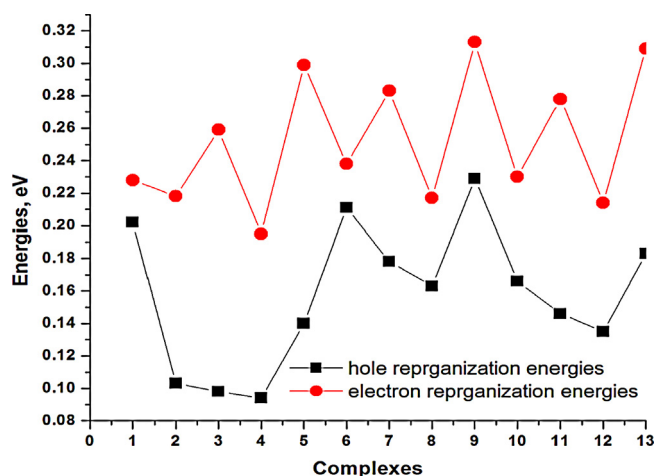


Fig. 6. The hole and electron reorganization energies of **1** and its derivatives (in eV) at the B3LYP/6-31G** level of theory.

the energies of the radical ion in its equilibrium geometry and in the neutral geometry.

The calculated hole reorganization energy of parent molecule is smaller than the electron one revealing that it might be better hole transfer material. The computed hole reorganization energies of all the new designed derivatives are smaller than the electron reorganization energies indicating these materials might be better hole transfer materials.

The significant effect toward lowering the hole reorganization energy has been observed in **2–5**, **7**, **8**, **10–13** compared to the parent molecule. Here we have compared the hole reorganization energies of all these materials with already well known materials to shed some light on the charge transfer properties of new designed derivatives. The hole reorganization energies of naphthalene, anthracene, tetracene and pentacene at B3LYP/6-31G** level of theory are 0.186, 0.138, 0.114, 0.098 eV, respectively. The hole reorganization energies of **2–5**, **7**, **8**, **10–13** are smaller than naphthalene. The hole reorganization energies of **2–4** and **12** are smaller than anthracene. The hole reorganization energies of **2–4** are smaller than tetracene. The hole reorganization energies of **3** and **4** are smaller than pentacene. It revealed that the new designed materials would be good hole transfer materials and **3** and **4** might be better/comparable to the pentacene which is commonly used hole transfer material. The computed electron reorganization energy of well know and commonly used electron transfer material (mer-Alq3) is 0.276 eV [45]. On the basis of reorganization energy, it seems that the electron mobility of **2–4**, **6**, **8**, **10–12** derivatives might be better/comparable with mer-Alq3. The frontier molecular orbitals (FMOs); highest occupied molecular orbitals (HOMOs) and lowest unoccupied molecular orbitals (LUMOs), ionization potential, electron affinity, reorganization energies suggesting that most of the designed derivatives would be better charge transfer materials, see Fig. 6.

4. Conclusions

The B3LYP functional is better than that of LC-BLYP to reproduce the experimental evidences. The comprehensible intra-molecular charge transfer has been observed from highest occupied molecular orbitals to the lowest unoccupied molecular orbitals. The HOMO–LUMO energy gap in new designed systems is smaller than the parent molecule. The stronger activating groups higher the HOMO energies. The trend to augment the HOMO energies on the base of strength of activating group is as OCH₃ > OH > CH₃ while contrary effect has been observed to lower the LUMO energies.

The trend to higher the HOMO energies on the base of strength of deactivating group is as $F > \text{COOH} > \text{CN} > \text{NO}_2$ while tendency to lower the LUMO energies is as $F > \text{COOH} > \text{CN} > \text{NO}_2$ both at ground and excited states. The smallest HOMO–LUMO energy gaps have been observed for those systems which have NO_2 or CN substituted groups. The stronger activating and deactivating groups reduced the energy gaps more efficiently. The lowered LUMO energy levels are revealing smaller injection barrier resulting improved electron injection in new designed derivatives. The stronger activating and deactivating groups enhance the dipole moment. The nitro-substituents showed fluorescence in visible and toward the longer wavelength, i.e., near IR revealing that these would be efficient sensitizers (dye sensitized solar cell materials). The IPa/IPv of new designed derivatives decreases while EAa/EAv increases compared to **1**. It showed that the injection barrier for hole and electron would decrease. The significant effect toward lowering the hole reorganization energy has been observed in **2–5**, **7**, **8**, **10–13** compared to the parent molecule. The hole reorganization energies of **2–4** and **12** are smaller than anthracene. The hole reorganization energies of **2–4** are smaller than tetracene. The hole reorganization energies of **3** and **4** are smaller than pentacene. The new designed materials would be good hole transfer materials and **3** and **4** might be better/comparable to the pentacene which is commonly used hole transfer material. The electron mobility of **2–4**, **6**, **8**, **10–12** derivatives might be better/comparable with *mer*-Alq3. Finally, it is expected that such type of materials might be better as light emitter field effect transistors. The new designed derivatives would be better light emitters; e.g. blue, green, orange, red and near IR).

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