

# SO<sub>2</sub> on TiO<sub>2</sub>(110) and Ti<sub>2</sub>O<sub>3</sub>(10 $\bar{1}$ 2) Nonpolar Surfaces: A DFT Study

Maurizio Casarin,<sup>\*,†</sup> Francesca Ferrigato,<sup>†</sup> Chiara Maccato,<sup>†</sup> and Andrea Vittadini<sup>‡</sup>

Dipartimento di Scienze Chimiche, Università di Padova, and Istituto di Scienze e Tecnologie Molecolari, C.N.R. di Padova, Padova, Italy

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Density functional molecular cluster calculations have been used to investigate the interaction of SO<sub>2</sub> with defect-free TiO<sub>2</sub>(110) and Ti<sub>2</sub>O<sub>3</sub>(10 $\bar{1}$ 2) surfaces. Adsorbate geometries and chemisorption enthalpies have been computed and discussed. Several local minima have been found for TiO<sub>2</sub>(110), but only one seems to be relevant for the catalytic conversion of SO<sub>2</sub> to S. In agreement with experiment, the bonding of SO<sub>2</sub> to Ti<sub>2</sub>O<sub>3</sub>(10 $\bar{1}$ 2) is much stronger than that on TiO<sub>2</sub>(110). Moreover, our results are consistent with the surface oxidation and the formation of strong Ti–O and Ti–S bonds. On both substrates, the bonding is characterized by a two-way electron flow involving a donation from the SO<sub>2</sub> HOMO into virtual orbitals of surface Lewis acid sites (L<sub>s</sub><sup>a</sup>), assisted by a back-donation from surface states into the SO<sub>2</sub> LUMO. However, the localization of surface states and the strength of back-donation are very different on the two surfaces. On TiO<sub>2</sub>(110), back-donation is weaker, and it involves unsaturated bridging O atoms, while on Ti<sub>2</sub>O<sub>3</sub>(10 $\bar{1}$ 2), it implies the L<sub>s</sub><sup>a</sup>-based valence band maximum and significantly weakens the S–O bond.

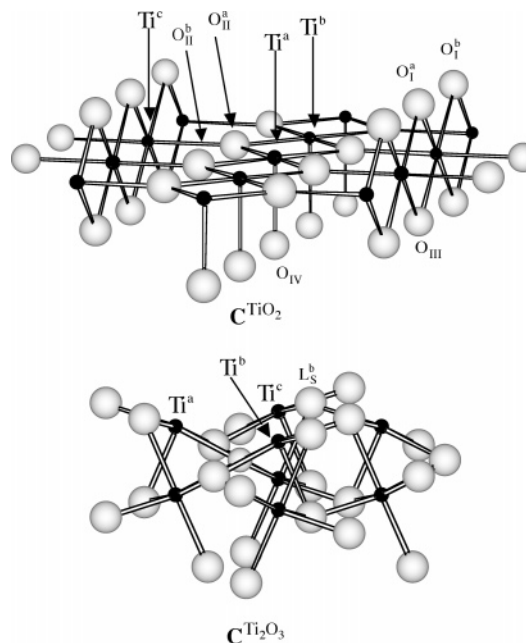
## 1. Introduction

TiO<sub>2</sub> and Ti<sub>2</sub>O<sub>3</sub> are probably the best-known among titanium oxides. In contrast to TiO<sub>2</sub>, Ti<sub>2</sub>O<sub>3</sub> is of little importance for catalytic applications. Nevertheless, the peculiar 3d<sup>1</sup> electronic configuration of its Lewis acid sites (L<sub>s</sub><sup>a</sup>) makes it appealing from a fundamental point of view, because it allows one to study the role played by partially occupied 3d atomic orbitals (AOs) in chemisorption.<sup>1</sup>

The investigation of the interaction of SO<sub>2</sub> with TiO<sub>2</sub> is particularly important, because this material is active in the sulfur recovery from natural, refinery, or coal gasification gases through the H<sub>2</sub>S + SO<sub>2</sub> Claus reaction.<sup>2,3</sup> Despite the fact that the interaction of SO<sub>2</sub> with rutile TiO<sub>2</sub>(110) has been extensively investigated from an experimental point of view,<sup>4</sup> the nature of the surface complexes is not yet completely clear.<sup>5</sup> Moreover, the few theoretical contributions<sup>5,6</sup> so far devoted to this argument have been mainly focused on the adsorption energetics rather than on the analysis of the adsorbate–substrate bonding scheme. In this regard, we know that, on the defect-free surface, SO<sub>2</sub> can be weakly bound both to L<sub>s</sub><sup>a</sup> and to Lewis base sites (the bridging oxygen atoms, hereafter L<sub>s</sub><sup>b</sup>), to give rise to SO<sub>2</sub> and SO<sub>3</sub>-like surface complexes.

As far as the adsorption on Ti<sub>2</sub>O<sub>3</sub> is concerned, the only paper so far dedicated to this topic<sup>7</sup> indicates a completely different behavior: SO<sub>2</sub> reacts vigorously with Ti<sub>2</sub>O<sub>3</sub>, its adsorption is dissociative in character, and it catalyzes the complete oxidation of the surface to TiO<sub>2</sub> and TiS<sub>2</sub>.

In our systematic study of the surface chemistry of metal oxides with molecular cluster models and density functional theory (DFT), we already investigated the interaction of a sulfur-containing molecule (H<sub>2</sub>S) with rutile TiO<sub>2</sub>(110)<sup>8</sup> and Ti<sub>2</sub>O<sub>3</sub>-(10 $\bar{1}$ 2).<sup>9</sup> In the present contribution, we extend our study to the adsorption of SO<sub>2</sub>.



**Figure 1.** Schematic representation of the employed clusters C<sup>TiO<sub>2</sub></sup> and C<sup>Ti<sub>2</sub>O<sub>3</sub></sup>. Saturators are not displayed for the sake of clarity.

## 2. Computational Details

Defect-free TiO<sub>2</sub>(110) has been modeled with a Ti<sub>13</sub>O<sub>29</sub> saturated cluster (C<sup>TiO<sub>2</sub></sup>, see Figure 1) centered on a five-fold-coordinated L<sub>s</sub><sup>a</sup> surface ion. C<sup>TiO<sub>2</sub></sup> is larger than the cluster adopted by some of us in ref 8 (Ti<sub>7</sub>O<sub>9</sub>) to model the chemisorption of CO, H<sub>2</sub>O, and H<sub>2</sub>S on rutile TiO<sub>2</sub>(110). This gives the opportunity of employing the same cluster to explore different coordinative modes. At variance to that, Ti<sub>2</sub>O<sub>3</sub>(10 $\bar{1}$ 2) has been mimicked by using the same Ti<sub>8</sub>O<sub>24</sub> cluster (C<sup>Ti<sub>2</sub>O<sub>3</sub></sup>, see Figure 1) already employed to investigate the chemisorption of CO, H<sub>2</sub>O, and H<sub>2</sub>S.<sup>3,9</sup> The experimental lattice constants (*a*<sup>TiO<sub>2</sub></sup> = *b*<sup>TiO<sub>2</sub></sup> = 4.587 Å, *c*<sup>TiO<sub>2</sub></sup> = 2.594 Å;<sup>10a</sup> *a*<sup>Ti<sub>2</sub>O<sub>3</sub></sup> = *b*<sup>Ti<sub>2</sub>O<sub>3</sub></sup> = 4.7570 Å, *c*<sup>Ti<sub>2</sub>O<sub>3</sub></sup> = 12.9877 Å<sup>10b</sup>) have been used for both systems.

\* To whom correspondence should be addressed. Phone/fax number ++39-049-8275164/8275161; E-mail maurizio.casarin@unipd.it.

<sup>†</sup> Dipartimento di Scienze Chimiche, Università di Padova.

<sup>‡</sup> Istituto di Scienze e Tecnologie Molecolari, C.N.R. di Padova.

Spurious surface states associated to dangling bonds (DBs) of not chemically complete titanium and oxygen cluster atoms have been eliminated by saturating them with pseudo-hydrogen/hydrogen atoms placed along the bond directions of the extended lattices. In this regard, it deserves to be mentioned that we avoided the embedding of the nonsaturated cluster in a point-charge (PC) array for several reasons. Among them, we emphasize that choosing the “best” values for PCs is almost a matter of taste: Typically adopted values range from the Mulliken charges to the substantially higher formal oxidation numbers. Moreover, in our previous study of the chemisorption of CO, H<sub>2</sub>O, and H<sub>2</sub>S on TiO<sub>2</sub>(110), we showed<sup>8</sup> that the negative effects due to the introduction of the Madelung field through the PC array are much heavier than those introduced by neglecting the field in the model adopting the pseudo-hydrogen saturators.

All the molecular cluster calculations have been run within DFT by using the *ADF 2002* package.<sup>11</sup> Optimized geometries and chemisorption enthalpies have been evaluated by employing generalized gradient corrections self-consistently included through the Becke–Perdew formula.<sup>12</sup> A triple- $\zeta$  Slater-type basis set has been used for Ti and adsorbate atoms, while a double- $\zeta$  basis set has been adopted for the host O atoms as well as for saturators. Inner cores of Ti (1s2s2p), S (1s2s2p), and O (1s) atoms have been treated by the frozen-core approximation.

The adsorption enthalpy ( $\Delta H$ ) has been analyzed in terms of adsorbate and substrate fragment orbitals through the use of the Ziegler’s extended transition-state method<sup>13</sup>

$$\Delta H = \Delta E_{\text{elstat}} + \Delta E_{\text{Pauli}} + \Delta E_{\text{int}} + \Delta E_{\text{prep}}$$

where  $\Delta E_{\text{elstat}}$  is the pure electrostatic interaction,  $\Delta E_{\text{Pauli}}$  is the destabilizing two-orbital-four-electron interaction between the occupied orbitals of the two interacting fragments ( $\Delta E_{\text{elstat}} + \Delta E_{\text{Pauli}} = \Delta E_{\text{steric}}$ ),  $\Delta E_{\text{int}}$  derives from the stabilizing interaction between occupied and empty orbitals of the fragments, and the last term,  $\Delta E_{\text{prep}}$ , is the energy required to relax the structure of the free fragments to the geometry of the final system. Adsorption enthalpies have been further corrected by taking into account the basis set superposition error (BSSE), which was estimated through the use of reference energies computed with ghost adsorbate and surface fragments.<sup>14</sup>

Information about the localization and the bonding/antibonding character of molecular orbitals (MOs) over a broad range of energies has been obtained by referring to density of states (DOS), partial DOS (PDOS), and crystal orbital overlap population (COOP) curves. These have been computed by weighting one-electron energy levels by their basis orbital percentage and by applying a 0.25 eV Lorentzian broadening.

### 3. Results and Discussion

We first checked the capability of  $\text{C}^{\text{TiO}_2}$  of reproducing the structural relaxation undergone by TiO<sub>2</sub>(110). For this purpose, we optimized the coordinates of Ti<sup>a</sup>, Ti<sup>b</sup>, Ti<sup>c</sup>, O<sub>I</sub><sup>a</sup>, O<sub>I</sub><sup>b</sup>, and O<sub>II</sub> (the atom labeling is reported in Figure 1) with the constraint of maintaining the  $C_{2v}$  symmetry of  $\text{C}^{\text{TiO}_2}$ . The obtained displacements compare quite well with literature experimental<sup>15</sup> and theoretical values;<sup>8,16</sup> moreover, the main features of the electronic structure of  $\text{C}^{\text{TiO}_2}$  are very similar to those obtained for the smaller Ti<sub>7</sub>O<sub>9</sub> cluster we adopted in ref 8.<sup>17,18</sup>

**3.1. The SO<sub>2</sub> Free Molecule.** Understanding the adsorption of SO<sub>2</sub> requires a detailed knowledge of its electronic structure. We briefly summarize the results obtained by running DFT calculations on the isolated molecule (see Table 1). Symmetry

**TABLE 1: Selected Molecular Properties of the SO<sub>2</sub> Molecule: Bond Length (BL), Bond Angle (BA), Dipole Moment ( $\mu$ ), Ionization Potential (IP), Binding Energy (BE), Vibrational Frequency ( $\nu$ )**

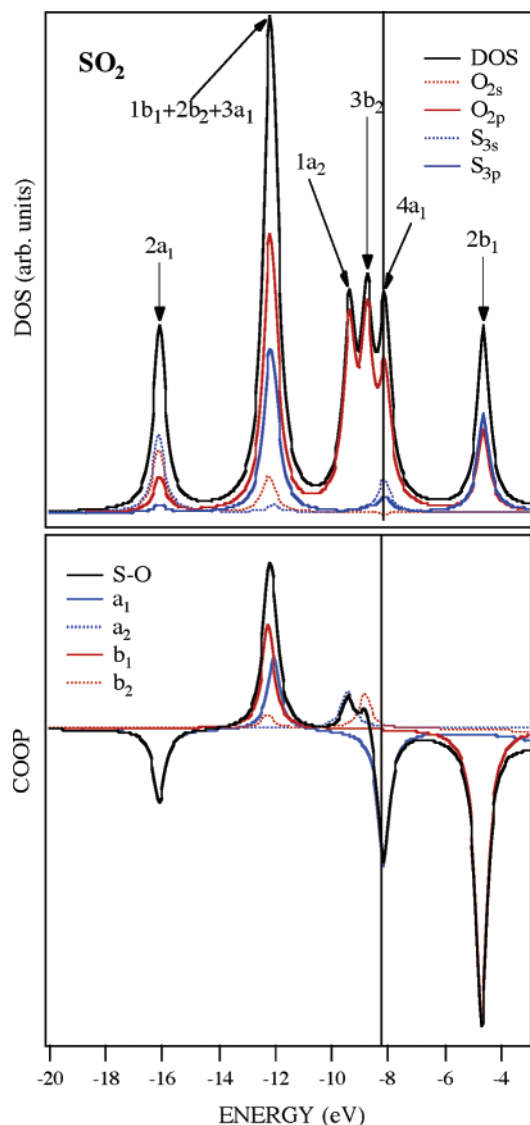
	theor	exptl
$r_{\text{S-O}}$ (Å)	1.465	1.432 <sup>a</sup>
O–S–O (°)	120.3	119.32 <sup>a</sup>
$\mu$ (D)	1.62	1.60 <sup>b</sup>
IP <sub>4a1</sub>	12.23	12.5 <sup>c</sup>
IP <sub>3b2</sub>	12.99	13.5 <sup>c</sup>
IP <sub>1a2</sub>	13.53	13.2 <sup>c</sup>
$\nu_1$ (cm <sup>-1</sup> )	1095.0	1151.4 <sup>d</sup>
$\nu_2$ (cm <sup>-1</sup> )	477.8	517.8 <sup>d</sup>
$\nu_3$ (cm <sup>-1</sup> )	1297.1	1360 <sup>d</sup>
BE (kcal/mol)	290.25	254.4 <sup>e</sup>
$Q_{\text{M}}^{\text{S}}$	1.04 <sup>f</sup>	
$Q_{\text{M}}^{\text{O}}$	−0.52 <sup>f</sup>	
$Q_{\text{H}}^{\text{S}}$	0.42 <sup>f</sup>	
$Q_{\text{H}}^{\text{O}}$	−0.21 <sup>f</sup>	

<sup>a</sup> ref 23. <sup>b</sup> ref 24. <sup>c</sup> ref 25. <sup>d</sup> ref 26. <sup>e</sup> ref 27. <sup>f</sup>  $Q_{\text{M}}^{\text{X}}$  and  $Q_{\text{H}}^{\text{X}}$  stand for Mulliken<sup>28</sup> and Hirschfeld<sup>29</sup> atomic charges.

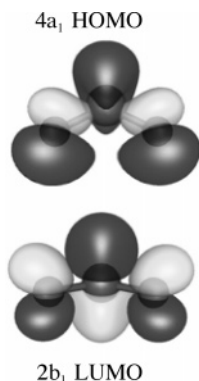
properties of the closed-shell SO<sub>2</sub> are described by the  $C_{2v}$  symmetry point group.<sup>20</sup> Its valence manifold (see Figure 2) includes a total of seven molecular orbitals MOs ((inner)  $1b_1^2 2b_2^2 3a_1^2 1a_2^2 3b_2^2 4a_1^2 / 2b_1^0$ ). Among them, both the  $4a_1$  HOMO and the  $2b_1$  LUMO<sup>21</sup> are antibonding in nature (see COOPs in Figures 2 and 3-D contour plots in Figure 3), even though the latter accounts for a stronger S–O repulsive interaction. Symmetry and overlap considerations suggest that SO<sub>2</sub> frontier orbitals will be the most perturbed upon chemisorption. However, as a consequence of their antibonding nature, the participation of the HOMO (LUMO) to the SO<sub>2</sub> → substrate (substrate → SO<sub>2</sub>) donation will strengthen (weaken) the S–O bond.

**3.2. SO<sub>2</sub> on the Defect-Free TiO<sub>2</sub>(110) Surface.** The interaction of SO<sub>2</sub> with TiO<sub>2</sub>(110) has been modeled by considering the different coordinative modes reported in Figure 4. Structures were optimized within the highest possible symmetries, that is,  $C_{2v}$  for SO<sub>2</sub> atop Ti<sup>a</sup> ( $a_1$  and  $a_2$ ),  $C_s$  for the SO<sub>3</sub>-like and  $\mu$ -SO<sub>2</sub> geometry, and  $C_1$  for the  $\mu$ -SO<sub>3</sub> and  $\eta^1$ -SO<sub>3</sub> coordination.

In Table 2, selected structural parameters, chemisorption enthalpies, and Hirschfeld/Mulliken atomic charges are reported, while different contributions to chemisorption enthalpy are displayed in Table 3. Data pertaining to the  $a_1$ -SO<sub>2</sub> geometry have not been included in the table, because this surface complex is computed to be highly unstable as a consequence of the strong repulsive interaction between the adsorbate O lone pairs ( $n$ ) and O<sub>II</sub>-based surface states. As a whole, the outcomes of our calculations indicate that the SO<sub>2</sub>–TiO<sub>2</sub>(110) interaction is quite weak, almost negligible for the SO<sub>3</sub>-like and  $\mu$ -SO<sub>3</sub> coordinations and a little stronger for the  $\mu$ -SO<sub>2</sub> and  $\eta^1$ -SO<sub>3</sub> arrangements. These results agree very well with supercell DFT calculations recently carried out by Zhang and Lindan (ZL);<sup>5</sup> moreover, it is noteworthy that our evaluation of the chemisorption enthalpy for the most stable arrangements ( $\mu$ -SO<sub>2</sub> and  $\eta^1$ -SO<sub>3</sub>) agrees quantitatively with those reported by ZL (−9.92 and −11.30 kcal/mol, respectively) for the same arrangements. Interestingly, despite the similarity of  $\Delta H^{\mu\text{-SO}_2}$  and  $\Delta H^{\eta^1\text{-SO}_3}$ , the two surface complexes seem to imply different bonding schemes. In fact, even though SO<sub>2</sub> behaves as an electron donor in both cases,<sup>30</sup> donation and back-donation mechanisms appear to compensate each other when the  $\eta^1$ -SO<sub>3</sub> complex is formed (see Table 2). In this regard, the information provided by data reported in Table 3 is particularly interesting. Among the

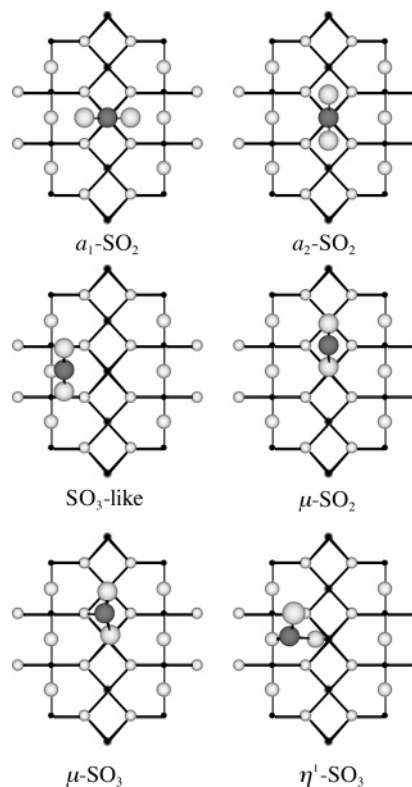


**Figure 2.** DOS (top) and corresponding COOPs (bottom) of the free  $\text{SO}_2$  molecule. Vertical bars represent the HOMO energy.



**Figure 3.** 3-D plots of the  $4a_1/2b_1$  HOMO/LUMO of the free  $\text{SO}_2$ . Dark and light surfaces correspond to the wave function values of  $0.04 \text{ e}^{1/2}/\text{\AA}^{3/2}$  and  $-0.04 \text{ e}^{1/2}/\text{\AA}^{3/2}$ , respectively.

coordinative modes we considered, those characterized by the strongest covalent interaction are the  $\mu\text{-SO}_3$  and  $\eta^1\text{-SO}_3$  ones. On the other hand,  $\Delta E_{\text{steric}}$  and  $\Delta E_{\text{prep}}$  corresponding to these geometries are also very high, and the final  $\Delta H$  is the result of a balance of different contributions. Moving to  $\mu\text{-SO}_2$ , we already mentioned that  $\Delta H^{\mu\text{-SO}_2}$  and  $\Delta H^{\eta^1\text{-SO}_3}$  are very similar. On the other hand, data reported in Table 3 clearly indicate



**Figure 4.** Schematic representation (top view) of the coordinative arrangements we explored for  $\text{SO}_2$  on the defect-free  $\text{TiO}_2(110)$ .

**TABLE 2: Selected Structural Parameters of  $\text{TiO}_2(110)\text{-SO}_2$  Surface Complexes. Adsorption Enthalpies and  $Q_{\text{H}}^{\text{SO}_2}/Q_{\text{M}}^{\text{SO}_2}$  Charges Are Also Included**

	$a_2\text{-SO}_2$	$\text{SO}_3\text{-like}$	$\mu\text{-SO}_2$	$\mu\text{-SO}_3$	$\eta^1\text{-SO}_3$
$r_{\text{S-O}}^{\text{I}} (\text{\AA})^b$	1.463	1.470	1.471	1.490	1.528
$r_{\text{S-O}}^{\text{O}} (\text{\AA})$	1.463	1.471	1.479	1.509	1.464
$r_{\text{Ti}^{\text{a}}\text{-S}} (\text{\AA})$	3.537				
$r_{\text{Ti}^{\text{a}}\text{-O}}^{\text{I}} (\text{\AA})$			2.526	2.374	2.142
$r_{\text{Ti}^{\text{b}}\text{-O}}^{\text{I}} (\text{\AA})$			2.418	2.280	
$r_{\text{O}}^{\text{a}}\text{-S} (\text{\AA})$		2.830		2.185	2.163
$\text{O}\text{S}\text{O} (^{\circ})$	120.4	118.0	119.2	115.0	112.6
$Q_{\text{H}}^{\text{SO}_2}/Q_{\text{M}}^{\text{SO}_2}$			0.38 (0.16)		0.14 (-0.05)
$\Delta H (\text{kcal/mol})$	0.06	-3.46	-9.53	-2.72	-10.98

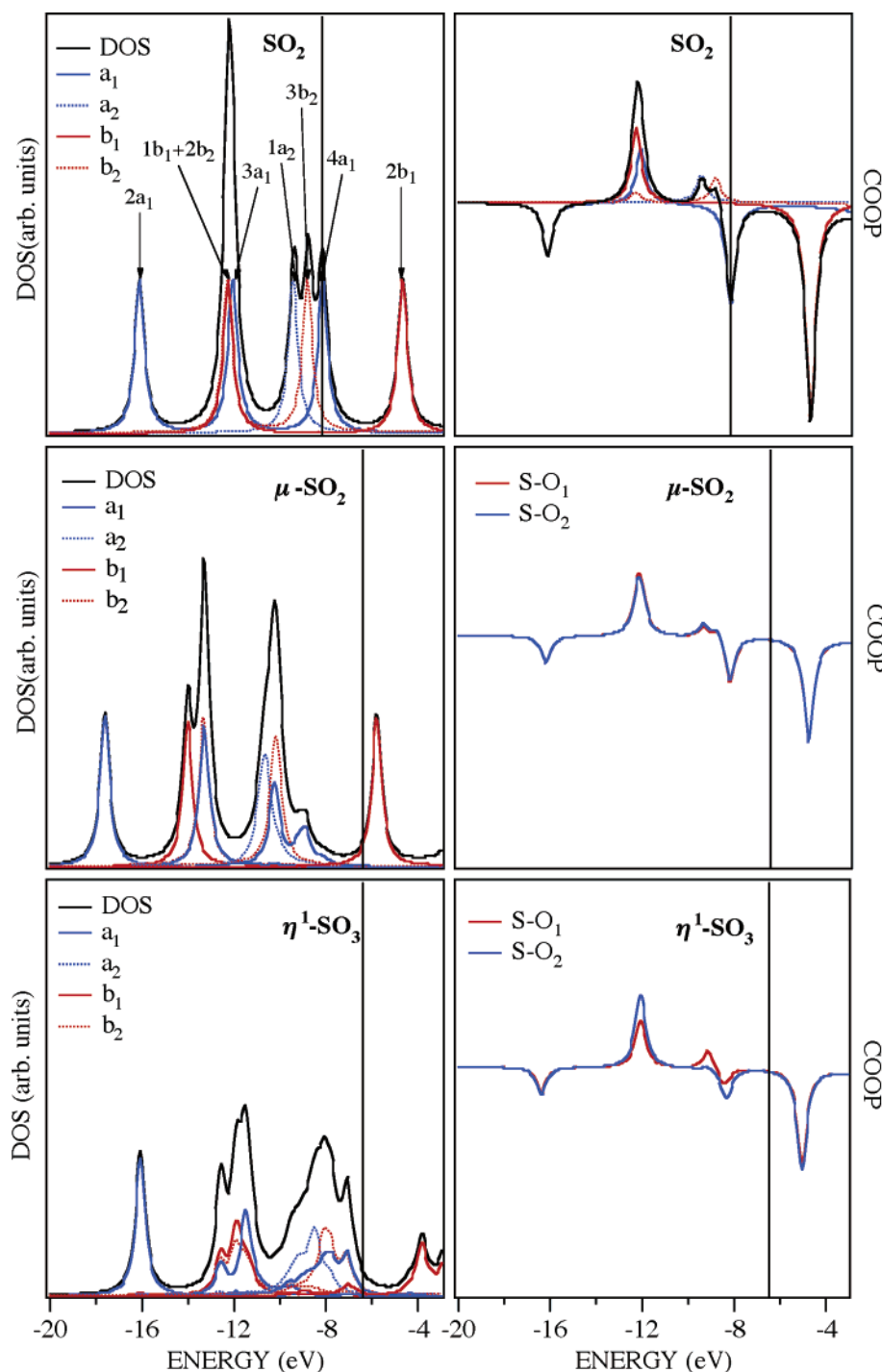
<sup>b</sup> The  $\text{O}^{\text{I}}$  corresponds to the  $\text{SO}_2$  O-atom bonded to the  $\text{L}_5^{\text{a}}$ .

**TABLE 3: Theoretical Adsorption Enthalpies (kcal/mol) Pertaining to  $\text{SO}_2$  on  $\text{TiO}_2(110)$  Decomposed by Means of the Ziegler Transition-State Method**

	$a_2\text{-SO}_2$	$\text{SO}_3\text{-like}$	$\mu\text{-SO}_2$	$\mu\text{-SO}_3$	$\eta^1\text{-SO}_3$
$\Delta E_{\text{steric}}$	2.69	1.14	6.49	59.85	48.57
$\Delta E_{\text{inter}}$	-4.13	-5.79	-19.74	-73.06	-75.96
$\Delta E_{\text{cluster}}^{\text{prep}}$	0.07	0.14	1.48	5.74	10.61
$\Delta E_{\text{prep}}^{\text{SO}_2}$	0.01	0.08	0.11	1.92	3.57
$\text{BSSE}_{\text{cluster}}$	0.87	0.61	1.57	1.97	1.50
$\text{BSSE}^{\text{SO}_2}$	0.55	0.36	0.56	0.86	0.73
$\Delta H$	0.06	-3.46	-9.53	-2.72	-10.98

that this is not due to a particularly strong covalent interaction between the substrate and the  $\text{SO}_2$  molecule symmetrically bridging two surface  $\text{L}_5^{\text{a}}$ , but rather, it is a consequence of the low  $\Delta E_{\text{steric}}$  and  $\Delta E_{\text{prep}}$  values.

Further insights into the  $\mu\text{-SO}_2$  and  $\eta^1\text{-SO}_3$  bonding schemes can be gained by referring to Figures 5 and 6. In the former, the  $\text{SO}_2$  PDOS and S-O COOPs for both coordinations are compared with corresponding DOS and COOPs of the free molecule, while in the latter,  $\text{Ti}^{\text{a}}\text{-SO}_2$ ,  $\text{O}^{\text{I}}\text{-SO}_2$ , and  $\text{O}^{\text{II}}\text{-SO}_2$



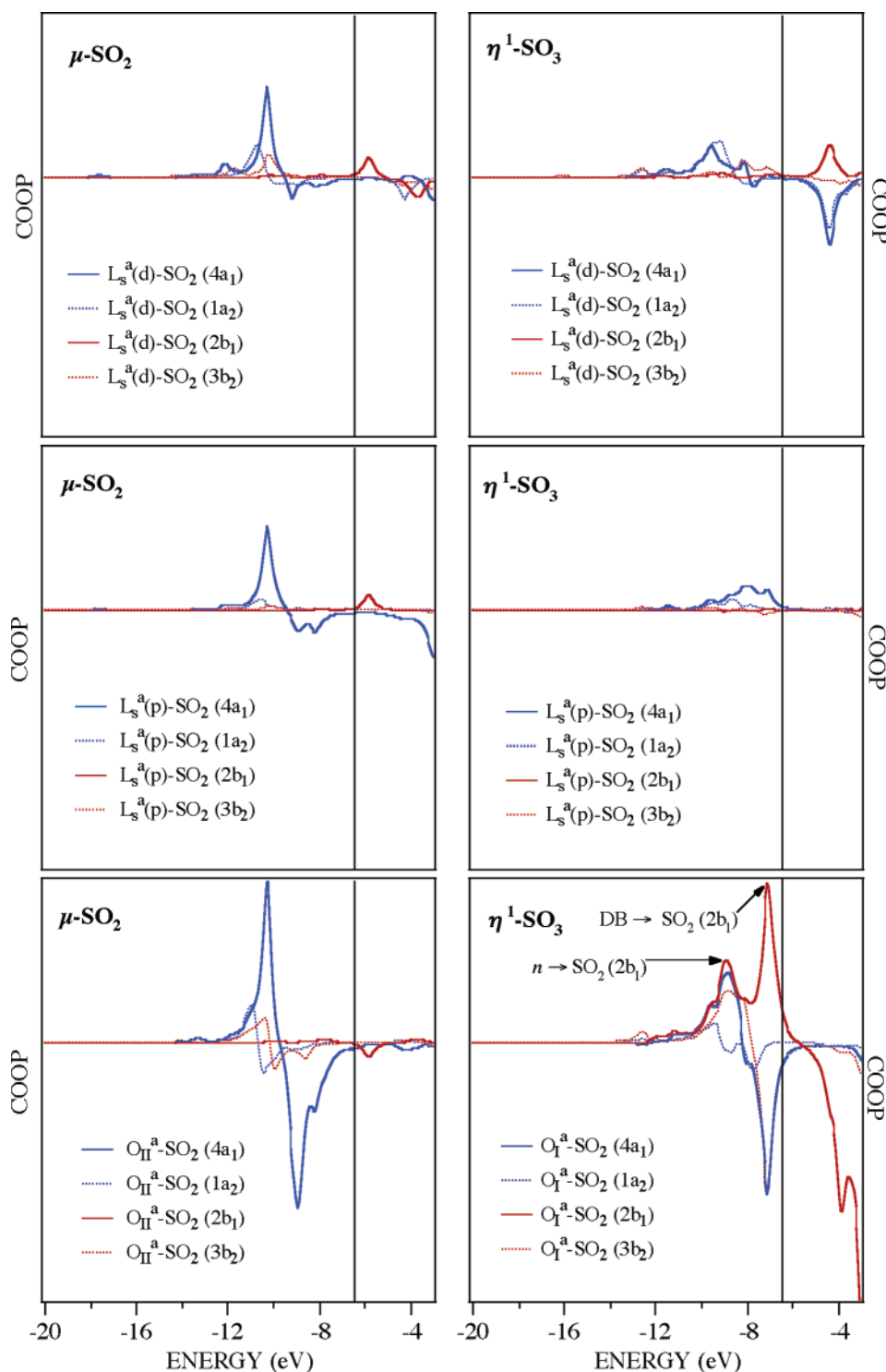
**Figure 5.** SO<sub>2</sub> PDOS and S–O COOPs corresponding to the μ-SO<sub>2</sub> and η<sup>1</sup>-SO<sub>3</sub> coordinative modes on the perfect TiO<sub>2</sub>(110) surface. Data pertaining to the free adsorbate are also included for comparison. Vertical bars represent the HOMO energy.

COOPs are displayed. The inspection of Figure 5 reveals that the SO<sub>2</sub> electronic structure is significantly perturbed upon chemisorption, independently of the surface complex geometry. More specifically, the MOs reminiscent of the SO<sub>2</sub> 4a<sub>1</sub>, 3b<sub>2</sub>, and 1a<sub>2</sub> levels are stabilized (destabilized) for the μ-SO<sub>2</sub> (η<sup>1</sup>-SO<sub>3</sub>) arrangement.<sup>31</sup> Moreover, Figure 6 testifies that the μ-SO<sub>2</sub> surface complex is characterized by an extensive charge transfer from the SO<sub>2</sub> HOMO (the weakly S–O antibonding 4a<sub>1</sub> level) into L<sub>s</sub><sup>a</sup> 3d and 4p virtual orbitals (L<sub>s</sub><sup>a</sup> 4s AOs participate to a negligible extent) and by the absence of any substrate → SO<sub>2</sub> back-donation.<sup>32</sup>

The bonding picture of the η<sup>1</sup>-SO<sub>3</sub> surface complex is more involved. The adsorbate–substrate interaction consists of a two-

way electron flow implying a donation from the SO<sub>2</sub> outermost MOs into L<sub>s</sub><sup>a</sup> virtual orbitals (again 3d and 4p AOs) assisted by a multicentered substrate → SO<sub>2</sub> back-donation involving a charge transfer from both the n<sup>O<sub>1</sub></sup> (parallel to the surface) and the DB<sup>O<sub>1</sub></sup> (perpendicular to the surface) into the SO<sub>2</sub> 2b<sub>1</sub> LUMO (see Figure 6). The significant participation of the SO<sub>2</sub>-based 2b<sub>1</sub> level (strongly S–O antibonding in nature) to the adsorbate–substrate interaction provides a rationale of the different electronic and structural perturbations undergone by the adsorbate upon chemisorption. In this regard, it is noteworthy that the weakening of a S–O bond, a necessary step of the catalytic conversion of SO<sub>2</sub> to S,<sup>2</sup> is favored only by the η<sup>1</sup>-SO<sub>3</sub> arrangement.





**Figure 6.**  $\text{Ti}^a\text{-SO}_2$ ,  $\text{O}_I\text{-SO}_2$ , and  $\text{O}_{II}\text{-SO}_2$  COOPs. The  $4s^{\text{Ti}^a}\text{-SO}_2$  has not been included because it is negligible. Vertical bars represent the HOMO energy.

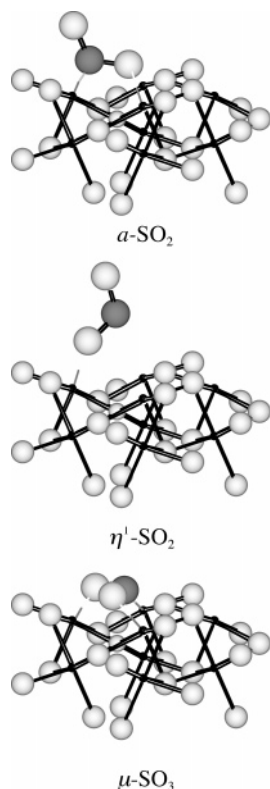
The analysis of the surface atomic displacements indicate that the strongest structural perturbations are pertinent to the  $\eta^1\text{-SO}_3$  geometry and that they are not limited to the atoms ( $\text{Ti}^a$ ,  $\text{O}_I^a$ ) directly involved in the adsorbate–substrate interaction.

As a whole, the obtained results compare quite well with experimental evidence indicating a molecular adsorption of  $\text{SO}_2$  on nearly perfect  $\text{TiO}_2(110)$  at low temperature through the involvement of surface  $\text{L}_s^a$  and  $\text{L}_s^{b,4c}$ .

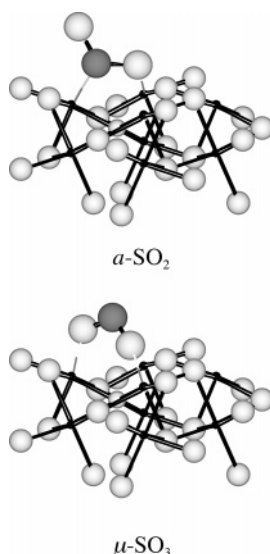
**3.3.  $\text{SO}_2$  on the Defect-Free  $\text{Ti}_2\text{O}_3(10\bar{1}2)$  Surface.** The interaction of  $\text{SO}_2$  with the defect-free  $\text{Ti}_2\text{O}_3(10\bar{1}2)$  surface has been modeled by considering three coordination geometries (see

Figure 7) and limiting the optimization procedure to the coordinates of the adsorbate atoms.<sup>3,9,33,34</sup>

In Table 4, selected structural parameters, chemisorption enthalpies, and  $Q_H/Q_M$  values are reported, while different contributions to chemisorption enthalpy are included in Table 5. Data pertaining to the  $\eta^1\text{-SO}_2$  starting coordination is missing, because it does not correspond to a local minimum. Theoretical results clearly indicate that, in agreement with experiment, the  $\text{SO}_2$ –substrate interaction is very different for  $\text{TiO}_2(110)$ <sup>4</sup> and  $\text{Ti}_2\text{O}_3(10\bar{1}2)$ <sup>7</sup> surfaces. In detail, (i) independently of the surface complex geometry,  $\text{SO}_2$  behaves as net electron acceptor (see

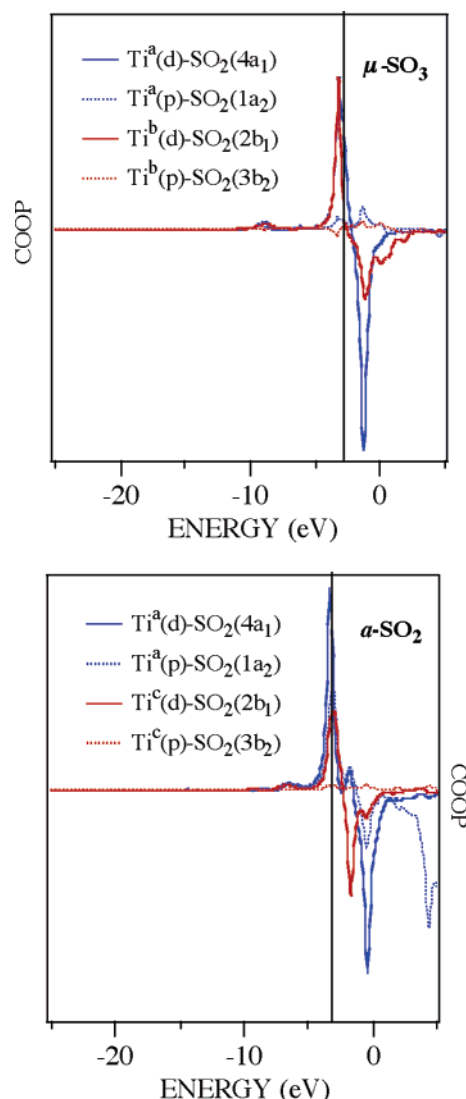


**Figure 7.** Schematic representation of the coordinative modes we explored for SO<sub>2</sub> on the defect-free Ti<sub>2</sub>O<sub>3</sub>(1012).



**Figure 8.** Schematic representation of the optimized  $\alpha$ -SO<sub>2</sub> and  $\mu$ -SO<sub>3</sub> coordinative modes on the unrelaxed Ti<sub>2</sub>O<sub>3</sub>(1012).

Table 4), (ii) the S–O bond lengths are dramatically lengthened upon chemisorption, (iii)  $L_s^a$ –O<sup>SO<sub>2</sub></sup> internuclear distances range between 1.924 and 1.982 Å (i.e., they are computed shorter than those of the Ti<sub>2</sub>O<sub>3</sub> bulk phase (2.0681 and 2.0245 Å)<sup>10b</sup> and very close to those pertaining to TiO<sub>2</sub> (1.949 and 1.969 Å)),<sup>10a</sup> (iv)  $L_s^a$ –S internuclear distance computed for the most stable arrangement ( $\mu$ -SO<sub>3</sub>) is 2.677 Å, a value slightly longer than the actual Ti–S bond lengths in the TiS<sub>2</sub> bulk phase (2.429 Å)<sup>35</sup> and in other nonstoichiometric titanium sulfides (2.332–2.544 Å),<sup>36–38</sup> and (v) the large  $\Delta H$  values and the corresponding  $\Delta E_{\text{inter}}$  and  $\Delta E_{\text{prep}}$  contributions are indicative of an incipient S–O bond breaking and of Ti–O and Ti–S bond formation. Interestingly, as already found for CO, H<sub>2</sub>O, and H<sub>2</sub>S on the



**Figure 9.** Ti<sup>a</sup>–SO<sub>2</sub>, Ti<sup>b</sup>–SO<sub>2</sub>, and Ti<sup>c</sup>–SO<sub>2</sub>, COOPs. Contributions from Ti-based 4s AOs have not been included because they are negligible. Vertical bars represent the HOMO energy.

**TABLE 4: Selected Bond Lengths (Å) and Bond Angles (°) of Ti<sub>2</sub>O<sub>3</sub>(1012)–SO<sub>2</sub> Surface Complexes. Adsorption Enthalpies (kcal/mol) and Charges Are Also Included<sup>a</sup>**

	$\alpha$ -SO <sub>2</sub>	$\mu$ -SO <sub>3</sub>
$r_{S-O^1}$	1.604	1.608
$r_{S-O^2}$	1.504	1.610
$r_{Ti^c-S}$	2.677	
$r_{Ti^c-O^1}$		1.924
$r_{Ti^c-O^2}$		1.952
$r_{Ti^c-O^1}$	1.982	
O <sup>SO</sup>	112.5	110.0
$Q_M^{SO_2}$	−0.18	−0.24
$Q_H^{SO_2}$	−0.47	−0.82
$\Delta H$ (kcal/mol)	−40.18	−63.45

<sup>a</sup> Atom labeling is reported in Figure 1.

same surface, the adsorbate–substrate interaction is dominated by covalent interactions (see Table 5) and, more specifically, by back-donation from the Ti<sub>2</sub>O<sub>3</sub> VBM into the 2b<sub>1</sub> SO<sub>2</sub> LUMO (see Figure 9). Such a participation of the Ti<sub>2</sub>O<sub>3</sub> VBM to the adsorbate–substrate interaction will certainly affect structural and electronic properties of the surface; on the other hand, any attempt to look into these perturbations is hampered by our frozen substrate approximation.<sup>39</sup>

**TABLE 5: Theoretical Adsorption Enthalpies (kcal/mol) Pertaining to SO<sub>2</sub> on Ti<sub>2</sub>O<sub>3</sub>(1012) Decomposed by Means of the Ziegler Transition-State Method**

	<i>a</i> -SO <sub>2</sub>	<i>μ</i> -SO <sub>3</sub>
$\Delta E_{\text{steric}}$	52.80	68.22
$\Delta E_{\text{inter}}$	-105.88	-154.86
$\Delta E_{\text{cluster}}^{\text{prep}}$		
$\Delta E_{\text{SO}_2}^{\text{prep}}$	11.04	21.02
BSSE <sub>cluster</sub>	1.07	1.36
BSSE <sub>SO<sub>2</sub></sub>	0.79	0.81
$\Delta H$	-40.18	-63.45

Despite their semiquantitative nature, theoretical results herein reported for SO<sub>2</sub> on Ti<sub>2</sub>O<sub>3</sub>(1012) testify to the crucial role played by the L<sub>s</sub><sup>a</sup> 3d electrons in chemisorption processes taking place on transition-metal oxide surfaces. This evidence confirms that, as pointed out by Barteau,<sup>40</sup> metal cations exposed at the surface of metal oxides have much in common with their counterparts in solution.

#### 4. Conclusion

Theoretical results herein reported demonstrate that small saturated clusters are able to predict the behavior of SO<sub>2</sub> on the clean, defect-free TiO<sub>2</sub>(110) surface. According to experiment and other theoretical outcomes, the adsorption is found to be molecular in nature and characterized by a small  $\Delta H$  (~10 kcal/mol). The most stable arrangements correspond to the *μ*-SO<sub>2</sub> and *η*<sup>1</sup>-SO<sub>3</sub> species, whose electronic structure is however very different. Interestingly, the latter surface complex could be of some relevance in the catalytic conversion of SO<sub>2</sub> to S. As far as the chemisorption of SO<sub>2</sub> on the Ti<sub>2</sub>O<sub>3</sub>(1012) surface, our results agree with experimental data in predicting a definitely stronger adsorbate–substrate interaction. Moreover, theoretical outcomes indicate that such an interaction is dominated by the back-donation from the L<sub>s</sub><sup>a</sup>-based VBM into the SO<sub>2</sub> 2b<sub>1</sub> LUMO.

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- (17) As far as the TiO<sub>2</sub>(110) electronic structure is concerned, we point out that the valence band maximum (VBM) is mainly localized on the surface L<sub>s</sub><sup>b</sup> (the bridging oxygen atoms), while the conduction band minimum (CBM) is mainly localized on 3d AOs of surface L<sub>s</sub><sup>a</sup>.
- (18) Although the electronic structure of C<sup>TiO<sub>3</sub></sup> has been thoroughly described in ref 3, it may be useful for the forthcoming discussion to herein summarize its main features. The outermost occupied MOs, representative of the Ti<sub>2</sub>O<sub>3</sub>(1012) VBM, are grouped in an energy range of 0.2 eV; they are all localized on the d<sub>z<sup>2</sup></sub> AO of the eight Ti atoms of C<sup>TiO<sub>3</sub></sup>, and all of them have a Ti–Ti bonding character. The top of the O 2p band lies 2.5 eV below the innermost of the Ti–Ti bonding MOs and, according to the results of the Henrich group,<sup>19</sup> it extends for ~6 eV. Furthermore, once again in agreement with the ultraviolet photoelectron spectroscopy (UPS) measurements of Smith and Henrich,<sup>19d</sup> the O 2p PDOS is characterized by a double-peaked structure. The higher-energy peak is due to nonbonding MOs (the threshold of the O 2p band is originated by the occupied L<sub>s</sub><sup>b</sup> dangling bonds), while the lower-energy one includes the bonding partners.
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- (31) On passing from the free SO<sub>2</sub> to *μ*-SO<sub>2</sub>, the shift toward lower energies of 4a<sub>1</sub>, 3b<sub>2</sub>, and 1a<sub>2</sub> MOs is not uniform. Corresponding  $|\Delta E|$ 's amount to 2.00, 1.32, and 1.14 eV, respectively.
- (32) Internuclear distances between L<sub>s</sub><sup>a</sup> and adsorbate O atoms (~2.46 Å) are quite close to the one we evaluated for H<sub>2</sub>O (2.41 Å)<sup>8</sup> on the same surface. Moreover, the slightly different L<sub>s</sub><sup>a</sup>–O values reported in Table 2 are due to the symmetry of the corresponding C<sup>TiO<sub>3</sub></sup>–SO<sub>2</sub> cluster: C<sub>s</sub>.
- (33) The starting geometries have been labeled as (i) *a*-SO<sub>2</sub>, with the adsorbate bonded to a surface L<sub>s</sub><sup>a</sup> through the S atom; (ii) *η*<sup>1</sup>-SO<sub>3</sub>, with the adsorbate bonded to a surface L<sub>s</sub><sup>a</sup> through one of its O atoms; (iii) *μ*-SO<sub>3</sub>, with the O atoms of the adsorbate positioned *a*-top two surface L<sub>s</sub><sup>a</sup> and the S species interacting with a surface L<sub>s</sub><sup>b</sup>.
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- (39) We already stressed in ref 3 that, even if C<sup>TiO<sub>3</sub></sup> reproduces well the electronic features of the clean Ti<sub>2</sub>O<sub>3</sub>(1012) surface, it is not well-suited to describe structural relaxation because of its asymmetry and limited size. Any attempt to get self-consistent field (SCF) convergence for larger clusters was not fruitful.<sup>34</sup>
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