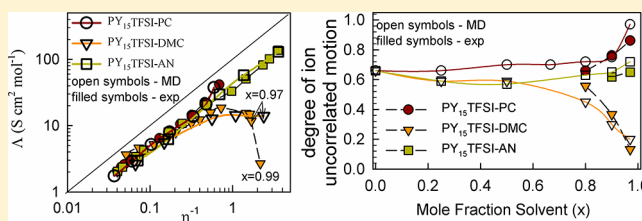


Influence of Solvent on Ion Aggregation and Transport in PY<sub>15</sub>TFSI Ionic Liquid–Aprotic Solvent MixturesOleg Borodin,<sup>\*,†</sup> Wesley A. Henderson,<sup>\*,‡</sup> Eric T. Fox,<sup>‡</sup> Marc Berman,<sup>§</sup> Mallory Gobet,<sup>§</sup> and Steve Greenbaum<sup>§</sup><sup>†</sup>Electrochemistry Branch, Sensor & Electron Devices Directorate, U.S. Army Research Laboratory, Adelphi, Maryland 20783, United States<sup>‡</sup>Ionic Liquids & Electrolytes for Energy Technologies (ILEET) Laboratory, Department of Chemical & Biomolecular Engineering, North Carolina State University, Raleigh, North Carolina 27695, United States<sup>§</sup>Department of Physics & Astronomy, Hunter College of the City University of New York, New York, New York 10065, United States

## S Supporting Information

**ABSTRACT:** Molecular dynamics (MD) simulations using a many-body polarizable APPLE&P force field have been performed on mixtures of the *N*-methyl-*N*-pentylpyrrolidinium bis(trifluoromethanesulfonyl)imide (PY<sub>15</sub>TFSI) ionic liquid (IL) with three molecular solvents: propylene carbonate (PC), dimethyl carbonate (DMC), and acetonitrile (AN). The MD simulations predict density, viscosity, and ionic conductivity values that agree well with the experimental results. In the solvent-rich regime, the ionic conductivity of the PY<sub>15</sub>TFSI–AN mixtures was found to be significantly higher than the conductivity of the corresponding –PC and –DMC mixtures, despite the similar viscosity values obtained from both the MD simulations and experiments for the –DMC and –AN mixtures. The significantly lower conductivity of the PY<sub>15</sub>TFSI–DMC mixtures, as compared to those for PY<sub>15</sub>TFSI–AN, in the solvent-rich regime was attributed to the more extensive ion aggregation observed for the –DMC mixtures. The PY<sub>15</sub>TFSI–DMC mixtures present an interesting case where the addition of the organic solvent to the IL results in an increase in the cation–anion correlations, in contrast to what is found for the mixtures with PC and AN, where ion motion became increasingly uncorrelated with addition of solvent. A combination of pfg-NMR and conductivity measurements confirmed the MD simulation predictions. Further insight into the molecular interactions and properties was also obtained using the MD simulations by examining the solvent distribution in the IL–solvent mixtures and the mixture excess properties.



## ■ INTRODUCTION

Ionic liquids (ILs) have attracted significant attention over the last two decades, as many of them have a negligible vapor pressure, exceptional thermal and electrochemical stability, favorable dissolution properties with many organic/inorganic compounds, and low flammability.<sup>1,2</sup> ILs, which may consist of a diverse variety of cations and anions, have been widely investigated for a variety of applications including biphasic systems for separation, solvents for synthetic and catalytic applications,<sup>3</sup> lubricants,<sup>4,5</sup> lithium batteries,<sup>6–8</sup> supercapacitors,<sup>9–11</sup> actuators,<sup>12,13</sup> reaction media<sup>14</sup> replacement of conventional solvents,<sup>2</sup> and active pharmaceutical ingredients.<sup>14</sup> Importantly, IL properties can be tailored for specific chemical or electrochemical applications by tuning the combination of cations and anions to achieve the desired thermodynamic, solvating, and transport properties, as well as safety. Their relatively low ionic conductivity and high cost, however, present significant challenges to IL adoption for many electrochemical applications. The introduction of low viscosity solvents has

therefore been explored in an effort to both improve the ion transport properties and reduce cost.

The availability of a large number of IL cations and anions, as well as organic solvents, presents tremendous opportunities for finding optimal cation/anion pairs and IL–solvent mixtures for the design of mixtures targeted toward specific applications. However, the screening of a large number of cation/anion and solvent combinations also presents an enormous challenge for product design, as the synthesis and characterization of a large number of ILs is expensive. Efficient and reliable predictive tools can speed up the development cycle not only by providing expedient predictions of properties for specific ILs and IL–solvent mixtures, but also by providing an improved fundamental understanding of the IL interactions and the data needed for the development of semiempirical structure–property relationship models. Molecular dynamics (MD)

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simulations are emerging as a powerful complementary option (to empirical correlations) for the reliable prediction of various properties of ILs. MD simulations also have the potential to provide insight needed for achieving a fundamental understanding of IL–solvent mixtures, as most structural, thermodynamic, and transport properties of ILs are readily accessible from simulations.<sup>15–26</sup>

This manuscript is a follow up to work recently reported consisting of a detailed experimental study of binary *N*-methyl-*N*-pentylpyrrolidinium bis(trifluoromethanesulfonyl)imide (PY<sub>15</sub>TFSI) mixtures with numerous aprotic solvents that revealed a discrepancy between the conductivity and viscosity behavior in the solvent-rich regime.<sup>27</sup> Specifically, despite the similar viscosity of the PY<sub>15</sub>TFSI mixtures with acetonitrile (AN) and dimethyl carbonate (DMC), the conductivity of the PY<sub>15</sub>TFSI–AN mixtures was significantly higher than the conductivity of the corresponding –DMC mixtures. On the other hand, similar conductivity values for the PY<sub>15</sub>TFSI–DMC and –propylene carbonate (PC) mixtures were observed despite the large difference in the viscosity of the mixtures. Preliminary MD simulations suggested that quite different ion aggregation behavior of the PY<sub>15</sub>TFSI ions when mixed with PC, DMC, or AN is responsible for the discrepancies noted between the IL–solvent conductivity and viscosity.<sup>27</sup> Chaban et al. also recently reported an unusually high conductivity for 1-alkyl-3-methylimidazolium tetrafluoroborate ILs when mixed with AN, based upon MD simulations.<sup>15</sup> Couadou et al. also reported that trimethylsulfonium TFSI and trimethylammonium TFSI have a significantly higher conductivity when mixed with AN relative to similar mixtures with PC or  $\gamma$ -butyrolactone (GBL).<sup>28</sup>

In this manuscript, we have utilized MD simulations to obtain a fundamental understanding of the relationship between the viscosity, conductivity, structural, and thermodynamic properties of PY<sub>15</sub>TFSI mixed with the PC, DMC, and AN solvents shown in Figure 1. PC has a significantly higher

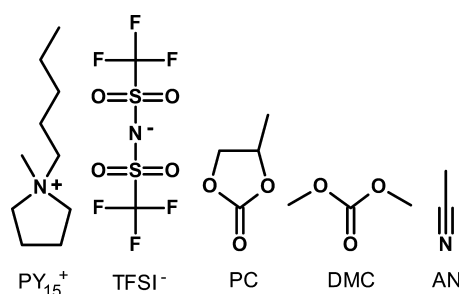


Figure 1. Structure of solvents and ions.

viscosity than DMC and AN, while DMC has a significantly lower dielectric constant and dipole moments than PC and AN, as shown in Table 1. These solvents allow us to investigate the influence of both solvent polarity and viscosity on the properties of the IL–solvent mixtures.

In this manuscript, the MD simulation predictions were validated against density, viscosity, and conductivity data. As ion association and correlated ion motion are central to the understanding of the IL–solvent conductivity behavior, it is especially important to ensure an accurate description of these factors for the IL–solvent mixtures. In order to further validate the MD simulation predictions for the ionic dissociation, a set of pfg-NMR self-diffusion coefficient measurements was

Table 1. Solvent Properties: Viscosity ( $\eta$ ) at 25 °C,<sup>27</sup> Dielectric Constant ( $\epsilon$ )<sup>27</sup> at 25 °C, and Gas-Phase 0 K Dipole Moment from MP2/aug-cc-pvTz Calculations for the Low Energy Conformers of DMC and PC

solvent	$\eta$ (mPa s)	$\epsilon$	dipole (D)
PC	2.51	65.0	5.58
DMC	0.59	3.1	0.33
AN	0.34	36.0	3.94

performed and combined with the ionic conductivity data to yield the degree of ion dynamic correlation that is often called ionicity in the literature.<sup>29</sup>

## EXPERIMENTAL AND COMPUTATIONAL SECTION

NMR experiments were run on a Varian direct drive 300 MHz spectrometer with a DOTY Z-gradient diffusion probe. All experiments were performed at 60 °C. A double stimulated echo pulse sequence was used to suppress convection effects, in conjunction with a saturation recovery sequence to shorten the data acquisition time. The data were collected and then fit to the equation to solve for the self-diffusion coefficient  $D$ :

$$I = I_0 \exp \left[ D \gamma^2 \delta^2 g^2 \left( T + \frac{4\delta}{3} + 2\delta_{sp} + 2\tau \right) \right] \quad (1)$$

Here  $I$  is the signal integral and  $I_0$  is a constant. The gradient values ( $g$ ) used ranged up to 55 G cm<sup>−1</sup>. Ring down times ( $\tau$ ) were 2 ms for all experiments.  $\delta$  gradient durations were 2 ms for <sup>1</sup>H and 4 ms for <sup>19</sup>F experiments.  $T$  diffusion times were 66 ms for <sup>1</sup>H and 100 ms for <sup>19</sup>F experiments.  $\delta_{sp}$ , the duration of the 80 G cm<sup>−1</sup> spoiler gradient, was set to 2 ms. Further details for the NMR experiments and NMR spectra are given in the Supporting Information. The obtained diffusion coefficients for the  $(1 - x_{\text{solv}})$  PY<sub>15</sub>TFSI +  $(x_{\text{solv}})$  solvent mixtures (with PC, DMC, or AN;  $x_{\text{solv}}$  is the solvent mole fraction) are given in Table 2.

Table 2. Diffusion Coefficients of Solvent ( $D_{\text{solv}}$ ), PY<sub>15</sub><sup>+</sup> ( $D_+$ ), and TFSI<sup>−</sup> ( $D_-$ ) for  $(1 - x_{\text{solv}})$  PY<sub>15</sub>TFSI +  $(x_{\text{solv}})$  S Mixtures (where S = AN, PC, and DMC) from NMR Measurements at 60 °C

solvent	$x_{\text{solv}}$	$D_{\text{solv}}$ (10 <sup>−9</sup> m <sup>2</sup> s <sup>−1</sup> )	$D_+$ (10 <sup>−9</sup> m <sup>2</sup> s <sup>−1</sup> )	$D_-$ (10 <sup>−9</sup> m <sup>2</sup> s <sup>−1</sup> )
PC	97	0.88	0.56	0.59
	90	0.62	0.44	0.43
	80	0.43	0.32	0.29
DMC	97	2.97	1.19	0.85
	90	1.93	0.64	0.74
	80	1.06	0.46	0.38
AN	97	4.57	2.39	2.27
	90	3.09	1.15	1.28

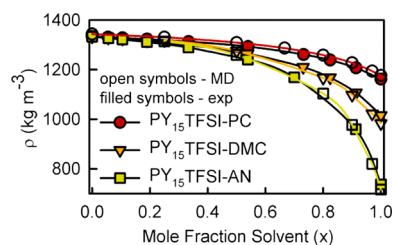
The previously developed many-body Atomistic Polarizable Potential for Liquids Electrolytes and Polymers (APPLE&P) force field was used for PY<sub>15</sub>TFSI and the PC, DMC, and AN solvents.<sup>30–33</sup> *Lucretius*, an MD simulation package that includes many-body polarization, was used for all the MD simulations. The following solvent mole fractions were simulated for all PY<sub>15</sub>TFSI–solvent mixtures:  $x_{\text{solv}} = 0, 0.25, 0.50, 0.80, 0.90, 0.97$ , and 1. An additional  $x_{\text{solv}} = 0.67$  concentration was simulated for PY<sub>15</sub>TFSI–PC mixtures. The PY<sub>15</sub><sup>+</sup> cations and TFSI<sup>−</sup> anions were dispersed in the simulation box at 120 °C by applying an additional repulsive

interaction between them. After the mixtures were equilibrated for 2–4 ns in the NPT ensemble at 120 °C, the simulation temperature was set to 60 °C and NPT runs were performed for 3.5–20 ns. The production runs were performed in the NVT ensemble for 8.2–44 ns depending on the IL concentration. The linear dimension of the simulation cell ranged from 38.4 to 51.7 Å. The Supporting Information lists the composition of the simulation cells, production and equilibration run length for each composition.

The Ewald summation method was used for the electrostatic interactions between permanent charges with permanent charges and permanent charges with induced dipole moments with  $k = 8^3$  vectors. The Thole screening parameter of 0.2, as described in the force field section, was used. The interaction between an induced dipole and a partial charge separated by three bonds was scaled by 0.8, providing an improved description of the electrostatic potential around the molecules. Multiple time step integration was employed with an inner time step of 0.5 fs (bonded interactions); a central time step of 1.5 fs for all nonbonded interactions within a truncation distance of 7.0 Å and an outer time step of 3.0 fs for all nonbonded interactions between 7.0 Å and the nonbonded truncation distance of 13–14 Å; as well as for the reciprocal part of Ewald. The induced dipole–induced dipole interactions were scaled from full interactions to zero using the fourth order polynomial in the interval [13.5 Å, 14 Å] for a 14 Å cutoff or over [12.5 Å, 13 Å] in the case of a 13 Å cutoff. A Nosé–Hoover thermostat and a barostat were used to control the temperature and pressure with the associated frequencies of  $10^{-2}$  and  $0.1 \times 10^{-4}$  fs. The stress tensor was saved every 9 fs for calculating stress, the tensor autocorrelation function, and viscosity, while the atomic coordinates were saved every 2 ps for postanalysis.

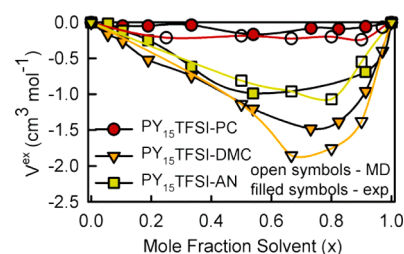
## RESULTS AND DISCUSSION

**Density.** The density of the PY<sub>15</sub>TFSI–solvent mixtures with PC, DMC, and AN solvents predicted from the MD simulations was found to be in good agreement with the measured data<sup>27</sup> for all of the studied solvent concentrations, as shown in Figure 2.



**Figure 2.** Density of PY<sub>15</sub>TFSI–solvent binary mixtures from MD simulations and experiments.

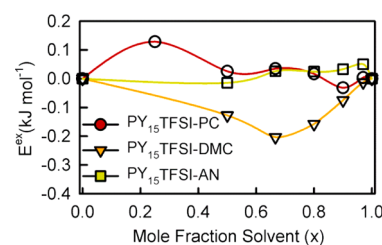
**Excess Volume.** The nonideality of the mixtures can be measured through the excess thermodynamic properties. We probed the excess volume and energy of mixing of the mixtures using the MD simulations. The excess volume for the PY<sub>15</sub>TFSI–solvent mixtures was negative for all of the compositions, as determined by both the MD simulations and experiments, as shown in Figure 3. The most negative values were observed for the PY<sub>15</sub>TFSI–DMC mixtures, while the least negative (almost zero) values were observed for the PY<sub>15</sub>TFSI–PC mixtures. It was previously found that large, relative to AN, cyclic molecules such as PC, EC, or GBL with a



**Figure 3.** Excess volume ( $V^{\text{ex}}$ ) for PY<sub>15</sub>TFSI–solvent binary mixtures from MD simulations and experiments.

large dipole moment had the smallest excess volumes when mixed with PY<sub>15</sub>TFSI, whereas less polar acyclic molecules tend to have the most negative excess volumes.<sup>27</sup> The magnitude of the observed excess volume for the PY<sub>15</sub>TFSI–AN mixtures is similar to the previously reported values for bmimBF<sub>4</sub>–AN mixtures.<sup>34</sup>

**Excess Energy.** The excess energy, calculated from MD simulations, is shown in Figure 4. The magnitude of the excess



**Figure 4.** Excess energy ( $E^{\text{ex}}$ ) for PY<sub>15</sub>TFSI–solvent binary mixtures from MD simulations.

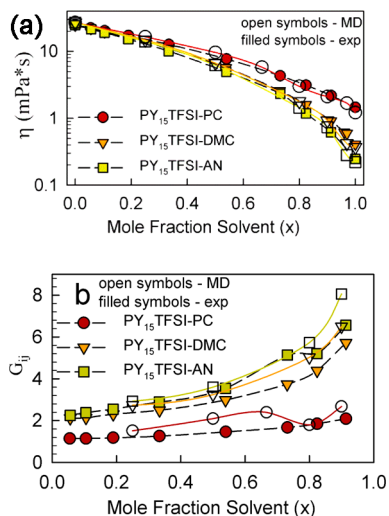
energy is small ( $<0.2 \text{ kJ mol}^{-1}$ ), indicating a similarity for the cation–anion and ion–solvent interactions. To put our predictions into perspective, we note that a positive excess enthalpy with a maximum of  $\sim 1\text{--}2 \text{ kJ mol}^{-1}$  was found for emimTriflate, emimBF<sub>4</sub>, and PY<sub>14</sub>BF<sub>4</sub> mixtures with water,<sup>35</sup> while mixtures of ILs having non-fluorinated ions, such as acetate or alkyl sulfate, with water resulted in more negative excess enthalpy values.<sup>35–37</sup> Previous experimental results for bmimBF<sub>4</sub> or bmimMeSO<sub>4</sub> mixtures with ethanol indicated maximum positive excess enthalpy values of  $\sim 2.5$  and  $1.3 \text{ kJ mol}^{-1}$ , respectively,<sup>38</sup> again following the trend that heavily fluorinated anions, such as BF<sub>4</sub><sup>−</sup>, mixed with polar solvents tend to yield more positive heats of mixing. While bmimBF<sub>4</sub> and bmimSO<sub>4</sub> mixtures with ethanol had positive heats of mixing, the same mixtures with nitromethane resulted in the negative heats of mixing.<sup>38</sup>

As the contribution of the ideal entropy of mixing to the free energy is always negative, the total entropic contribution is unlikely to be positive, yielding negative free energies of mixing values. Thus, our predicted very small energies of mixing are consistent with the experimentally observed mixing of IL solvents over the entire concentration range at 60 °C. Interestingly, there is a correlation between the excess volume and energy results shown in Figures 3 and 4—specifically, both properties are the most negative for PY<sub>15</sub>TFSI–DMC and the most positive for PY<sub>15</sub>TFSI–PC.

**Viscosity.** The viscosity of the PY<sub>15</sub>TFSI–solvent mixtures was calculated using the Einstein relation, as discussed in the Supporting Information. The PY<sub>15</sub>TFSI–solvent mixture viscosity values from the MD simulations and experiments<sup>27</sup>



are shown in Figure 5a. These data indicate that the viscosity of the PY<sub>15</sub>TFSI–PC mixtures is higher than that for the



**Figure 5.** (a) Viscosity and (b) viscosity interaction parameter ( $G_{ij}$ ) of PY<sub>15</sub>TFSI–solvent mixtures from MD simulations and experiments.

PY<sub>15</sub>TFSI–DMC and PY<sub>15</sub>TFSI–AN mixtures, as would be expected from the order of increasing viscosity for the pure solvents. The excess viscosity is shown in the Supporting Information as Figure S7. It is negative for all of the mixtures and follows the order PY<sub>15</sub>TFSI–AN < PY<sub>15</sub>TFSI–DMC < PY<sub>15</sub>TFSI–PC both for the experimental results and MD simulations. A large negative excess viscosity was also previously reported for bmimPF<sub>6</sub>–AN mixtures.<sup>39</sup>

One of the most widely used equations (eq 2) for predicting binary mixture viscosity is due to Grunberg and Nissan:<sup>40</sup>

$$\ln \eta_m = x_1 \ln \eta_1 + x_2 \ln \eta_2 + x_1 x_2 G_{ij} \quad (2)$$

where  $x_1$  and  $x_2$  are the molar fractions of components 1 and 2;  $\eta_1$ ,  $\eta_2$ , and  $\eta_m$  are the viscosity of component 1, component 2, and the binary mixture, respectively; and  $G_{ij}$  is an interaction parameter that denotes the extent of nonideality of the system which is a function of the composition.  $G_{ij}$  was suggested to be related to the interchange energy arising from the increase in lattice activation energy due to juxtaposition of molecule 1 into the lattice of molecule 2.<sup>41</sup> If one replaces  $\eta_n$  in eq 2 with  $\eta_n v_n$ , where  $v_n$  is the molar volume of the  $n$ th component, then the term  $x_1 x_2 G_{ij}$  is related to the molar Gibbs free energy of activation for the flow process in the Eyring absolute rate theory.<sup>42</sup> The  $G_{ij}$  parameter is plotted in Figure 5b. It is smallest for the PY<sub>15</sub>TFSI–PC mixtures and largest for the –AN mixtures. This ordering for the  $G_{ij}$  parameter weakly correlates with the order of excess volume shown in Figure 3. The increase of  $G_{ij}$  with increasing solvent mole fraction ( $x_{\text{solvent}}$ ) is attributed to the larger number of the ion–solvent contacts as the ion's first coordination shell gets filled upon addition of solvent up to  $x_{\text{solvent}} = 0.90$ . This is consistent with the largest magnitude of excess volume being around  $x_{\text{solvent}} = 0.80$  (see Figure 3).

**Ion Transport.** Ion self-diffusion coefficients ( $D$ ) and ionic conductivity ( $\kappa$ ) was extracted using the Einstein relations shown in eqs 3 and 4:

$$D_i = \lim_{t \rightarrow \infty} \frac{1}{6} \sum_i^N \langle ([\mathbf{R}_i(t) - \mathbf{R}_i(0)])^2 \rangle \quad (3)$$

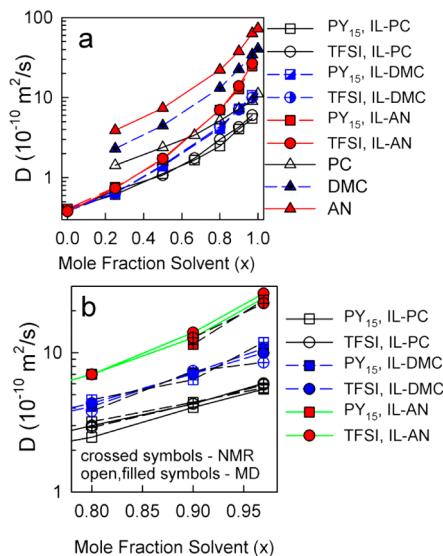
$$\kappa = \lim_{t \rightarrow \infty} \frac{e^2}{6tVk_B T} \sum_{i,j} z_i z_j \langle ([\mathbf{R}_i(t) - \mathbf{R}_i(0)])([\mathbf{R}_j(t) - \mathbf{R}_j(0)]) \rangle \quad (4)$$

where  $e$  is the electron charge,  $V$  is the volume of the simulation box,  $k_B$  is Boltzmann's constant,  $T$  is the temperature,  $t$  is time,  $z_i$  and  $z_j$  are the charges over ions  $i$  and  $j$  in electrons,  $\mathbf{R}_i(t)$  is the displacement of ion  $i$  during time  $t$ ,  $\langle \rangle$  denotes the ensemble average, and  $N$  is the number of diffusing species in eq 3 and the sum of cations plus anions in the MD simulation cell in eq 4. Due to the finite size of the simulation cell, long-range hydrodynamic interactions restrict diffusion.<sup>43</sup> The leading order finite size correction (FSC) to the self-diffusion coefficient (change to  $\Delta D^{\text{FSC}}$ ) was found to be inversely proportional to the MD simulation box and is given by eq 5:<sup>43</sup>

$$\Delta D^{\text{FSC}} = \frac{2.837 k_B T}{6\pi\eta L} \quad (5)$$

where  $k_B$  is the Boltzmann constant,  $T$  is temperature,  $L$  is a linear dimension of the simulation periodic cell, and  $\eta$  is viscosity. The magnitude of FSC was 8–15% for solvents and 14–50% for ions with the largest values observed in dilute PY<sub>15</sub>TFSI–DMC solutions. The FSC was also applied to ionic conductivity by increasing it by the same amount as the ion diffusion coefficients were increased.

The ion diffusion coefficients predicted from the MD simulations are shown in Figure 6a, with excellent agreement

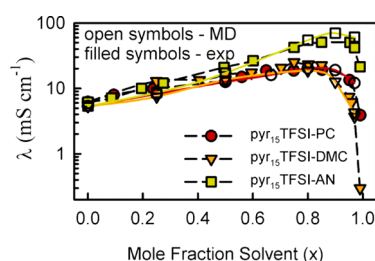


**Figure 6.** (a) Solvent and ion self-diffusion coefficients from MD simulations and (b) comparison of the ion self-diffusion coefficient from NMR and MD simulations.

between the MD simulation predictions and pfg-NMR measurements shown in Figure 6b. Solvent self-diffusion coefficients follow the inverse of the dependence of the viscosity, as expected, but the viscosity decreases faster upon solvent addition than the solvent diffusion coefficient increases, especially for the PY<sub>15</sub>TFSI–AN mixtures. The AN and DMC diffusion coefficients increase faster than the PC diffusion coefficient upon solvent addition, which is in line with the

viscosity behavior shown in Figure 5a. For the PY<sub>15</sub>TFSI–DMC mixtures, the ion and solvent diffusion coefficients increase at the same rate, while the diffusion coefficients increase at a higher rate for the ions than for the solvents upon addition of solvent for the PY<sub>15</sub>TFSI–AN and –PC mixtures, as shown in Figure S6 in the Supporting Information. At  $x_{\text{solv}} = 0.25$ , the ratios of the average ion diffusion coefficient to the solvent diffusion coefficient are 0.46, 0.29, and 0.19 for PC, DMC, and AN, respectively. This correlates with the sizes of the solvent molecules, which follow the order PC > DMC > AN. The order of the ratios of ion to solvent diffusion coefficients changes in the solvent-rich phase  $x_{\text{solv}} = 0.97$ , where the ratios are 0.64, 0.31, and 0.42 from the MD simulations and 0.65, 0.40, and 0.50 from the pfg-NMR experiments for PC, DMC, and AN, respectively. The fact that the addition of solvent increases the ion diffusion coefficients relative to the solvents for the –PC and –AN containing mixtures, but not for the PY<sub>15</sub>TFSI–DMC mixtures is consistent with the PC and AN solvents effectively separating the ions, thus weakening the ion–ion interactions and increasing the ion diffusion coefficients, whereas DMC is less effective at the ion separation.

The ionic conductivity for the mixtures from the MD simulations and experiments is shown in Figure 7.<sup>27</sup> The MD

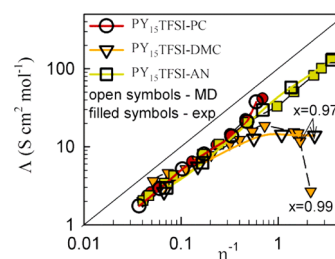


**Figure 7.** Ionic conductivity from MD simulations and from experiments.<sup>27</sup>

simulations not only accurately predict the correct position of the conductivity maximum, but also the magnitude of the conductivity for all of the concentrations. Notably, the conductivity of the PY<sub>15</sub>TFSI–solvent mixtures does not follow the viscosity behavior shown in Figure 5. Specifically, from the mixture viscosity values, the conductivities of the PY<sub>15</sub>TFSI–AN and –DMC mixtures are expected to be similar in magnitude with the conductivity of the –AN mixtures being only slightly higher than that for the –DMC mixtures. Instead, the conductivities of the PY<sub>15</sub>TFSI–DMC and –PC mixtures are similar and significantly lower than the conductivity of the –AN mixtures.

In order to further examine the relationship between the conductivity and viscosity, the molar conductivity was plotted against fluidity (inverse of viscosity) in the Walden plot shown in Figure 8. The deviation from the ideal behavior line is commonly associated with the correlated ion motion and ion pair/aggregate formation. The PY<sub>15</sub>TFSI–PC and –AN curves are parallel to the ideal line, indicating that the correlation of ion motion does not change upon addition of the solvent. The addition of DMC to PY<sub>15</sub>TFSI, however, results in an increasing deviation from the ideal behavior as the solvent concentration increases past  $x_{\text{solv}} = 0.80$ .

The degree of ion uncorrelated motion ( $\alpha_d$ ) can be directly accessed from the MD simulations and a combination of pfg-NMR and conductivity measurements. It is also often called ionicity and is given by eqs 6 and 7:

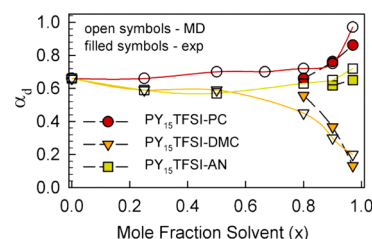


**Figure 8.** Walden plot showing inverse viscosity ( $\eta^{-1}$ ) vs molar conductivity ( $\Lambda$ ).

$$\alpha_d = \frac{\kappa}{\kappa_{\text{uncorr}}} \quad (6)$$

$$\kappa_{\text{uncorr}} = \frac{e^2}{V k_B T} (n_+ D_+ + n_- D_-) \quad (7)$$

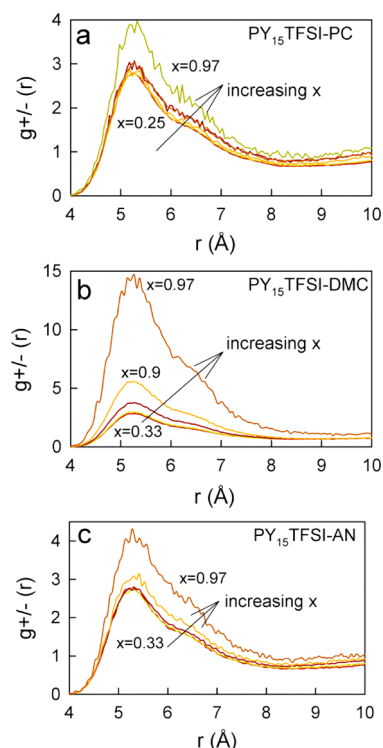
where  $e$  is the electron charge,  $V$  is the volume of the sample,  $k_B$  is Boltzmann's constant,  $T$  is the temperature, and  $n_+$  and  $n_-$  are the number of cations or anions, respectively.  $\kappa_{\text{uncorr}}$  is the “ideal” conductivity that would be realized if ion motion were uncorrelated. The degree of ion uncorrelated motion from the MD simulations and experiments is shown in Figure 9. This



**Figure 9.** The degree of ion uncorrelated motion from MD simulations and experiments.

indicates that the MD simulations accurately capture the degree of ion correlations extracted from the combined experimental measurements. The degree of ion uncorrelated motion is essentially unchanged for the PY<sub>15</sub>TFSI–AN mixtures over the entire concentration range studied. In the solvent-rich regime ( $x_{\text{solv}} > 0.80$ ), the ionic motion in the PY<sub>15</sub>TFSI–PC mixtures becomes slightly less correlated, while the correlation between the cation and anion diffusion dramatically increases for the PY<sub>15</sub>TFSI–DMC mixtures, as reflected by the significant drop in  $\alpha_d$  noted in Figure 9. A dramatic decrease of  $\alpha_d$  was also observed by Tokuda et al.<sup>44</sup> for IL mixtures with a nonpolar solvent: *N,N*-diethyl-*N*-methoxyethyl-*N*-methyl ammonium DEMEBF<sub>4</sub> and DEMETFSI with 1,2-dichloroethane (DCE).

**Structural Properties.** Radial distribution functions (RDFs) of the PY<sub>15</sub>TFSI–solvent mixtures were examined in order to better understand the relationship between the structural and transport properties. Figure 10 shows the center of mass cation–anion RDFs as a function of concentration. For the PY<sub>15</sub>TFSI–PC and –AN mixtures, the first peak of the cation–anion RDFs is essentially unchanged from  $x_{\text{solv}} = 0.33$  to 0.90, followed by a small increase for  $x_{\text{solv}}$  from 0.90 to 0.97. The behavior of the PY<sub>15</sub>TFSI–DMC mixtures, however, is quantitatively different, with a significant increase in the first cation–anion peak noticeable for  $x_{\text{solv}} = 0.80$ . This is followed by a dramatic increase up to 15 upon a further increase of the DMC mole fraction up to  $x_{\text{solv}} = 0.97$ . The cation–anion RDF



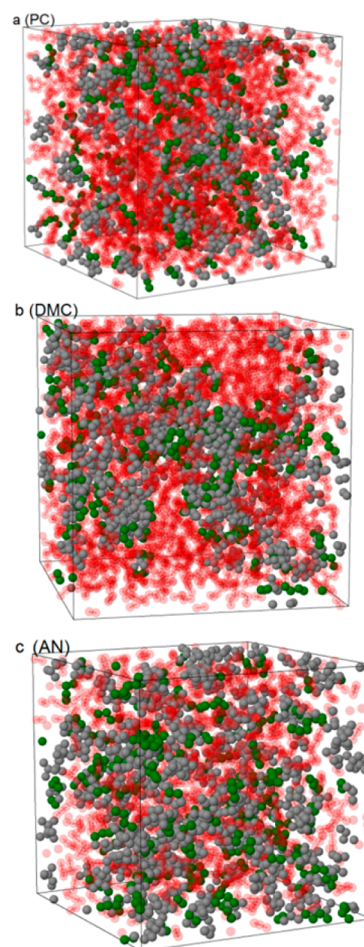
**Figure 10.** The cation–anion center of mass radial distribution functions from MD simulations.

behavior therefore correlates well with the increase in the correlation of the cation–anion motion observed for the PY<sub>15</sub>TFSI–DMC mixtures, with little change in  $\alpha_d$  for the other mixtures with –PC and –AN (see Figure 9).

Indeed, ion aggregation is quite pronounced in the simulation snapshot for the PY<sub>15</sub>TFSI–DMC mixture shown in Figure 11b, with nanodomains of ion- and solvent-rich regions observed. In contrast, the ions are well dispersed in the PY<sub>15</sub>TFSI–PC and –AN simulation cells shown in Figure 11a and c, which is consistent with the RDF behavior shown in Figure 10. We also examined the solvent aggregation in the IL-rich regime ( $x_{\text{solv}} = 0.33$ ) and found that the magnitude of the first RDF peak is similar for the three solvents, having a value of 1.6–1.8. This indicates some moderate aggregation of solvent in the PY<sub>15</sub>TFSI, as shown in more detail in the Supporting Information in Figure S10.

## CONCLUSIONS

A detailed analysis of mixtures of the PY<sub>15</sub>TFSI IL with PC, DMC, and AN was performed using MD simulations, as well as viscosity, conductivity, density, and pfg-NMR experimental measurements. The PY<sub>15</sub>TFSI–AN mixtures were found to have a much higher conductivity than the other two mixtures, despite the similar viscosity to the PY<sub>15</sub>TFSI–DMC mixtures. The conductivity of the PY<sub>15</sub>TFSI–PC mixtures was found to be similar to the PY<sub>15</sub>TFSI–DMC mixtures, despite the much higher viscosity for the former. The viscosity and conductivity behavior was reconciled by the finding that significant ion aggregation occurs in the solvent-rich region for the PY<sub>15</sub>TFSI–DMC mixtures, with the degree of ion uncorrelated motion (ionicity) dramatically decreasing from 0.6 to below 0.2 due to the formation of large transient aggregates as the mixture is further diluted with DMC. Thus, when the IL is mixed with low dielectric constant solvents, such as DMC, significant



**Figure 11.** Snapshots of MD simulation box for  $x_{\text{solv}} = 0.90$  for  $(1 - x_{\text{solv}})$  PY<sub>15</sub>TFSI–( $x_{\text{solv}}$ ) solvent with (a) PC, (b) DMC, and (c) AN. The solvent is transparent red, while the ions are opaque gray for the charged TFSI<sup>−</sup> anion and N<sup>+</sup>(CH<sub>3</sub>)(CH<sub>2</sub>) portion of the PY<sub>15</sub><sup>+</sup> cation and green for the neutral –C<sub>4</sub>H<sub>9</sub>– and –C<sub>2</sub>H<sub>4</sub>– portions of the PY<sub>15</sub><sup>+</sup> cation.

cation–anion aggregation is expected to occur in the solvent-rich regime. The most aggregated PY<sub>15</sub>TFSI–DMC mixture counterintuitively had the most negative excess volume and excess energy amongst the three studied mixtures. These results indicate that the most negative excess volume and excess energy do not necessarily imply the strongest ion–solvent interactions and cannot be directly linked to ion dissociation because the excess properties are measured relative to the pure solvent and IL. For example, the most negative excess volume for the strongly aggregating PY<sub>15</sub>TFSI–DMC mixture indicates that the ion aggregates fit DMC better than the solvated ions fit PC and AN.

The solvent diffusion coefficients increased to a lesser degree with increasing solvent concentration as compared to the increase of the mixture fluidity (inverse of viscosity) as the ion diffusion was slower than solvent diffusion. For the PY<sub>15</sub>TFSI–PC and –AN mixtures, the ion diffusion increased at a higher rate than the solvent diffusion upon solvent addition due to an increase in the ion dissociation. In contrast, the ratio of the ion diffusion to solvent diffusion was essentially unchanged for the PY<sub>15</sub>TFSI–DMC mixtures due to pronounced ion aggregation in the solvent-rich regime.



## ■ ASSOCIATED CONTENT

## ■ Supporting Information

Additional details for the NMR measurements; snapshots of the MD simulations for the PY<sub>15</sub>TFSI mixtures with PC, DMC, and AN; and details regarding the extraction of the viscosity data from the MD simulations. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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## Notes

The authors declare no competing financial interest.

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