# Atomistic Simulation of Nafion Membrane. 2. Dynamics of Water Molecules and **Hydronium Ions**

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We have performed a detailed and comprehensive analysis of the dynamics of water molecules and hydronium ions in hydrated Nafion using classical molecular dynamics simulations with the DREIDING force field. In addition to calculating diffusion coefficients as a function of hydration level, we have also determined mean residence time of H<sub>2</sub>O molecules and H<sub>3</sub>O<sup>+</sup> ions in the first solvation shell of SO<sub>3</sub><sup>-</sup> groups. The diffusion coefficient of H<sub>2</sub>O molecules increases with increasing hydration level and is in good agreement with experiment. The mean residence time of H<sub>2</sub>O molecules decreases with increasing membrane hydration from 1 ns at a low hydration level to 75 ps at the highest hydration level studied. These dynamical changes are related to the changes in membrane nanostructure reported in the first part of this work. Our results provide insights into slow proton dynamics observed in neutron scattering experiments and are consistent with the Gebel model of Nafion structure.

### 1. Introduction

Polymer electrolyte membrane (PEM) fuel cells convert chemical energy to electrical energy with high efficiency and minimal pollution. Their widespread deployment as part of the proposed hydrogen economy<sup>1</sup> can reduce dependence on fossil fuels, environmental pollution, and greenhouse gas emissions. Currently, PEM fuel cells use perfluorosulfonic acid (PFSA)based membranes, such as Nafion developed by DuPont. The desired membrane properties for this application include high proton conductivity at low hydration levels, thermal, mechanical, and chemical stability, durability under prolonged operation, and low cost. None of the existing membranes meet all the requirements and there is a pressing need to develop novel membranes based on a molecular level understanding of membrane chemistry and nanostructure.

In PFSA membranes, water molecules play a multifaceted role as a shuttle for proton transport, reaction product, coolant, and an impediment to reactant transport at high concentrations.<sup>2</sup> The dynamics of water molecules confined within nanoscale regions of PFSA membranes, especially at low hydration levels, is remarkably different from that of water molecules in bulk water.<sup>3</sup> A fundamental understanding of the relationship between membrane nanostructure and the dynamics of water molecules is needed for the development of efficient, reliable, and costeffective membranes to advance PEM fuel cell technology.

In an effort to interpret experimental data and understand the relationship between Nafion membrane nanostructure and the water pores and channels in Nafion membrane. In the first part of our work.<sup>8</sup> we examined the effect of hydration on membrane nanostructure and showed that water molecules and hydronium ions are strongly bound to sulfonate groups for  $\lambda$ 

less than 7. We showed that multiple sulfonate groups surround the hydronium ion at low hydration.<sup>8,9</sup> In the present report,

been extensively studied by experiment. Kreuer et al. 10 have reviewed transport in proton conductors for fuel cell applications. They have highlighted the differences in transport mechanisms with changes in the level of hydration, which is the ratio of water molecules per sulfonic acid group in Nafion and is typically represented by  $\lambda$ . Kreuer et al. 10 have stated that the mechanism of proton conduction is structural diffusion at high levels of hydration ( $\lambda > 10$ ) and vehicular transport (H<sub>3</sub>O<sup>+</sup>) at intermediate and low levels of hydration. In this report, the terms structural diffusion and structural transport of protons are used to refer to the Grotthuss mechanism that has been discussed in great detail by Marx et al.<sup>11</sup>

Zawodzinski et al.4 determined proton diffusion coefficients from conductivity measurements and water diffusion coefficients from nuclear magnetic resonance (NMR) measurements in Nafion at various levels of hydration ( $\lambda$  ranging from 2 to 22). The two diffusion coefficients are comparable at low  $\lambda$  (~3), while the proton diffusion coefficient diverges to higher values as  $\lambda$  increases. The authors have inferred from these results that structural diffusion becomes more significant at higher  $\lambda$  due to more bulklike behavior of water, which is consistent with the conclusions of Kreuer et al.<sup>10</sup>

Pivovar and Pivovar<sup>5</sup> examined hydrated Nafion 117 using quasi-elastic neutron scattering (QENS) and observed two distinct regimes of dynamical behavior. The notation 117 refers to equivalent weight of 1100 g/mol of SO<sub>3</sub><sup>-</sup> group and a membrane thickness of 0.007 in. Pivovar and Pivovar<sup>5</sup> fitted their data using two different models based on continuous diffusion within a sphere and unconstrained random diffusion and extracted water residence times of the order of 10 ps. The diffusion coefficients of water molecules obtained by Pivovar

molecular transport, we have performed classical molecular dynamics (MD) simulations using an all-atom force field at levels of membrane hydration chosen to match experiment.<sup>4-7</sup> We use the term nanostructure to refer to the structure within

we compute the dynamical properties of water molecules and hydronium ions in Nafion and relate them to the structural changes reported previously.8 Proton and small molecule transport in Nafion membrane has

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and Pivovar<sup>5</sup> using QENS are considerably greater than those reported by Zawodzinski et al.<sup>4</sup> using the NMR technique.

The water molecule residence time reported by Pivovar and Pivovar<sup>5</sup> decreases rapidly with increasing  $\lambda$  up to  $\lambda=6$  and then levels off, indicating increasing water mobility with increasing  $\lambda$ . This is consistent with an infrared spectroscopy study<sup>6</sup> that reported an abrupt structural change in hydrated Nafion around  $\lambda=5$ . The simulation results discussed in the first part of this work<sup>8</sup> provide an explanation for this structural change in terms of coordination of hydronium ions by multiple sulfonate groups.

In contrast to the work of Pivovar and Pivovar,<sup>5</sup> a recent QENS study<sup>7</sup> of water dynamics in hydrated Nafion 112 has used a single theoretical model based on Gaussian statistics valid over the whole neutron momentum transfer (O) range to fit the data. This recent study<sup>7</sup> has concluded that there are two distinct populations of diffusing proton. Both have characteristic length scales of 2-4 Å, but the characteristic time scales differ by a factor of 50 at all hydration levels. The slow dynamics has a time scale of  $\sim 500$  ps at low  $\lambda$  and  $\sim 160$  ps at high  $\lambda$ . The slow proton population has been identified with a long-lived hydronium moiety with the implication that the hydronium ion lifetime in hydrated Nafion is of the order of 1 ns, which is in contrast to the case in bulk water where the hydronium ion lifetime is  $\sim 1$  ps. This supports a slow diffusion mechanism where the hydronium ion moves as a well-defined entity (vehicular transport) at all hydration levels in Nafion.

Proton transport and water diffusion in hydrated Nafion have also been examined by numerous molecular modeling studies<sup>8,9,12-17</sup> and their findings have been recently summarized by Elliott and Paddison.<sup>18</sup> Petersen and Voth<sup>12</sup> have found the vehicular and structural transport of protons to be of similar magnitude but strongly negatively correlated in their study of hydrated Nafion using a self-consistent multistate empirical valence bond (SCI-MS-EVB) method. They concluded that sulfonate ions act as proton traps and limit hydronium diffusion. A previous study<sup>13</sup> using a statistical mechanical model had shown that proton mobility is highest in the central region of hydrated Nafion pores well-separated from sulfonate groups.

Urata et al.<sup>15</sup> used a united-atom force field to study water dynamics in Nafion membrane for  $\lambda=2.8,\,5.9,\,13.3,\,$  and 35.4. Their calculated water diffusion coefficients were smaller than experimental values. Their results showed that water molecules located 2–4 Å away from the nearest sulfonate group had a shorter residence time than those located 4–6 or 6–8 Å away at  $\lambda=13.3$  and 35.4, which is at odds with the findings of Paul and Paddison.<sup>13</sup> Moreover, the residence time of water molecules beyond 6 Å was found to increase with increasing level of hydration, which is in disagreement with the trends seen in QENS experiments.<sup>5,7</sup> These disagreements may have arisen from the limitations imposed by the use of united-atom force fields. Moreover, a recent work<sup>19</sup> has discussed the need to rigorously define the mean residence time of molecules.

In the present work, we have systematically determined the mean residence time of water molecules as a function of Nafion membrane hydration level. The paper is organized as follows: in section 2, we discuss the details of the simulation including the calculation of mean residence time of water molecules; in section 3, we discuss the diffusion of water molecules and hydronium ions in light of experimental findings; in section 4, we summarize our findings.

### 2. Details of the Simulation

**2.1. Simulation Cell and Force Field.** We performed classical MD simulations of hydrated Nafion with all-atom force

Figure 1. Chemical structure of Nafion.

fields. The chemical formula of a Nafion chain in the present work is shown in Figure 1. Each chain of Nafion has ten hydrophilic  $SO_3^-$ -terminated pendants spaced evenly by seven nonpolar  $-CF_2-CF_2-$  monomers that form a hydrophobic backbone, referred to as a dispersed sequence by Jang et al. We assume all the sulfonate groups to be ionized based on the findings of Paddison and Elliott<sup>20</sup> for  $\lambda \geq 3$ . Four Nafion chains of 682 atoms each were contained in an orthorhombic simulation box with 40  $H_3O^+$  ions and from 0 to 760  $H_2O$  molecules ( $H_2O$  wt % = 1.5–23.9) with periodic boundary conditions imposed. The simulated cells correspond to  $\lambda$  of 1, 3, 5, 7, 9, 11, 13.5, and 20, respectively. Since ab initio calculations<sup>20</sup> have shown that the proton is unlikely to transfer from the sulfonic acid group to water for  $\lambda = 1$ , this case was simulated mainly to establish a classical benchmark.

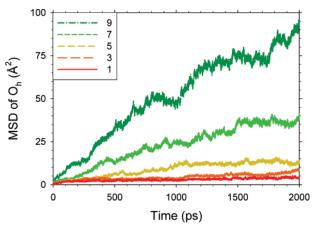
We performed all our simulations using the DL\_POLY 2.16 code<sup>21</sup> and the DREIDING force field<sup>14,22</sup> for Nafion and F3C force field<sup>23</sup> for H<sub>2</sub>O and H<sub>3</sub>O<sup>+</sup>. The details of the implementation of these potentials such as cutoff distances have been discussed previously.<sup>8</sup> Following energy minimization and annealing for each  $\lambda$ , which have been described in detail elsewhere,<sup>8</sup> we equilibrated the system for 250 ps at 300 K using NPT MD simulation. The resulting system density varied from 1.76 g/cm<sup>3</sup> for  $\lambda = 1$  to 1.62 g/cm<sup>3</sup> for  $\lambda = 20$  and is in good agreement with corresponding experimental values.<sup>24</sup> Following equilibration, we performed NVT MD simulations at 300 K for 2 ns for each value of  $\lambda$  and saved the configurations at 0.2 ps intervals.

**2.2. Diffusion Coefficients.** We determined the diffusion coefficients, D, for the water oxygen  $(O_w)$  and hydronium oxygen  $(O_h)$  for various  $\lambda$  from the mean square displacement  $(MSD = \langle |r(t) - r(0)|^2 \rangle)$  as follows:

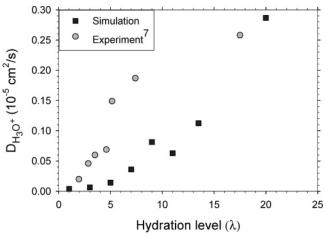
$$D = \frac{\langle |r(t) - r(0)|^2 \rangle}{6t} \tag{1}$$

We evaluated the MSD of  $O_w$  and  $O_h$  at 0.2 ps intervals and determined D over the time period from 1 to 2 ns.

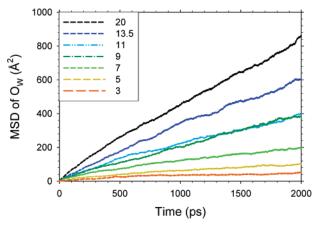
**2.3. Mean Residence Time.** We have calculated mean residence times ( $\tau_{\rm MR}$ ) of H<sub>2</sub>O molecules and H<sub>3</sub>O<sup>+</sup> ions in the first hydration shell around the SO<sub>3</sub><sup>-</sup> groups to relate hydrated Nafion membrane nanostructure to dynamics of small molecules and ions. An approach commonly used in the literature is to define  $\tau_{\rm MR}$  of a molecule as the time span it stays in the solvation shell based on the idea of persisting coordination.<sup>25–27</sup> To reduce problems that arise in the determination of  $\tau_{\rm MR}$  when a molecule makes frequent transits across the solvation shell boundary, excursions outside the solvation shell for a brief period,  $t^*$ , are allowed. The use of an arbitrarily chosen parameter,  $t^*$ , is considered a crude device to avoid elaborate bookkeeping<sup>25</sup> and its value has ranged from  $0^{27}$  to 2 ps.<sup>25</sup> Schröder et al.<sup>19</sup> have



**Figure 2.** MSD of hydronium oxygen in Nafion at low hydration levels  $(\lambda)$  indicated by the legend.



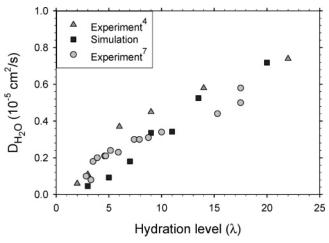
**Figure 3.** Diffusion coefficient of  $H_3O^+$  ions at various hydration levels in Nafion from the present simulation (squares) and QENS experiment (circles).



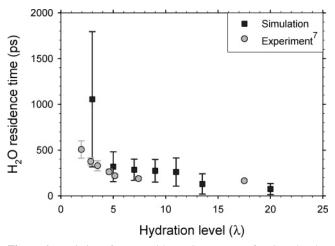
**Figure 4.** MSD of water oxygen in Nafion for hydration levels  $(\lambda)$  indicated by the legend.

argued that the above approach is unsatisfactory because it yields  $\tau_{MR}$  values that are dependent on the radius of the solvation shell and the frequency of recorded trajectories.

Following the approach of Brunne et al.,<sup>28</sup> we have calculated  $\tau_{\rm MR}$  of H<sub>2</sub>O molecules and H<sub>3</sub>O<sup>+</sup> ions around a given SO<sub>3</sub><sup>-</sup> group from 2000 configurations at 1 ps intervals. Consider a Boolean variable  $v_i(t)$  that is 1 if the ith H<sub>2</sub>O molecule is in the



**Figure 5.** Variation of diffusion coefficient of H<sub>2</sub>O molecules with hydration level in Nafion from the present simulation (squares), NMR experiment (triangles), and QENS experiment (circles).



**Figure 6.** Variation of mean residence time (MRT) of H<sub>2</sub>O molecules with hydration level in Nafion from the present simulation (squares) and MRT associated with slow dynamics observed in QENS experiment<sup>8</sup> (circles).

solvation shell of a  $SO_3^-$  group at time t and 0 otherwise. The correlation function of this Boolean variable can be written as

$$f(t_k) = \sum_{i=1}^{M-k} \sum_{i=1}^{N} v_i(t_j) \ v_i(t_j + t_k)$$
 (2)

 $t_k = k\Delta t$  is the offset time, where k = 0, 1, 2, ..., 1800 and  $\Delta t = 1$  ps. The inner summation is over N H<sub>2</sub>O molecules, while the outer summation is over M-k time origins,  $t_j$ , such that  $t_j = t_{j-1} + \Delta t$ ,  $t_1 = 1$  ps, and M = 2000. The mean residence time,  $\tau_{\rm MR}$ , is obtained by fitting the correlation function to an exponential function, such that

$$f(t) = A \exp(-t/\tau_{\rm MR}) \tag{3}$$

The same approach was used for  $H_3O^+$  ions around a given  $SO_3^-$  group. The cutoff distance for the first solvation shell around a  $SO_3^-$  group was 4.3 Å for both  $H_2O$  molecules and  $H_3O^+$  ions. This distance is consistent with our definition of bound water in the first part of this work.<sup>8</sup>

# 3. Results and Discussion

3.1. Diffusion Coefficients of  $H_3O^+$  Ions and  $H_2O$  Molecules. Figure 2 shows the MSD of  $O_h$  as a function of time

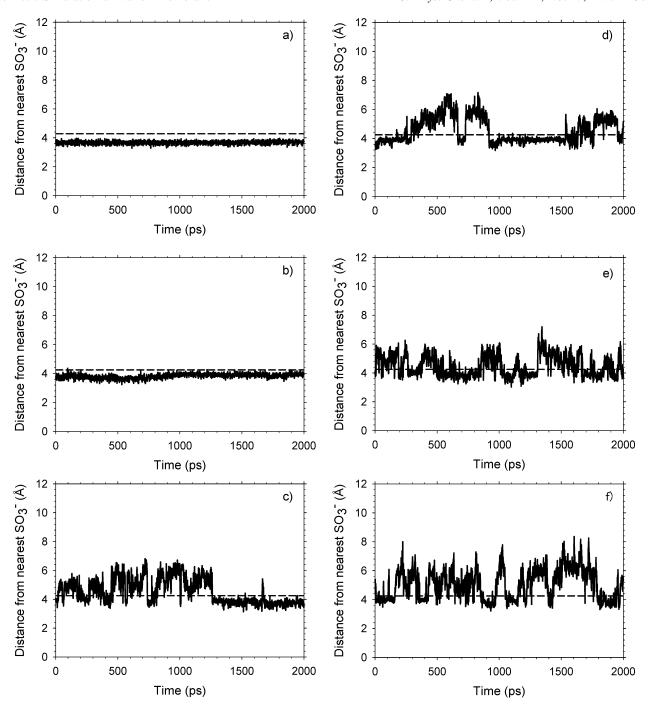
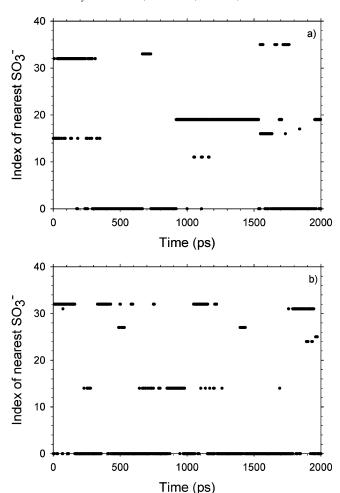


Figure 7. Distance from the nearest  $SO_3^-$  group for a  $H_3O^+$  ion for  $\lambda$  values of (a) 1, (b) 3, (c) 5, (d) 7, (e) 9, and (f) 20. The horizontal line represents the radius of the first solvation shell.

for low hydration levels. We have plotted the data for  $\lambda$  values up to 9 to highlight the change in  $H_3O^+$  ion dynamics around  $\lambda=5$ . Our simulation reveals that there is a significant increase in the mobility of  $O_h$  for  $\lambda=7$  compared to that for  $\lambda\leq 5$ . We obtained the diffusion coefficient from the plot of MSD as a function of time according to eq 1 by averaging over 100 values sampled between 1 and 2 ns. Figure 3 shows the vehicular diffusion coefficient of the classical  $H_3O^+$  ion from the present simulation at various hydration levels and the local diffusion coefficient of slow protons, identified as  $H_3O^+$  ions, from QENS experiment. The hydration levels in the present study were chosen to cover the range of values examined by experiment, the simulation and interpretation of experimental results. The simulated values of  $D_{H_3O^+}$  are smaller than the corresponding experimental values by a factor

of 2-5 except at  $\lambda \sim 20$ , where the agreement with experiment is good. This discrepancy reflects the limitation of classical models in probing hydronium dynamics. However, general trends in the experimental data are reproduced by the simulations. For instance, there is an abrupt increase in the diffusion of hydronium ions at  $\lambda = 5$  as shown by experiment.<sup>7</sup> This is related to changes in the hydration number and bridging configurations of  $H_3O^+$  ions between  $SO_3^-$  groups observed in the first part of this work<sup>8</sup> in agreement with the findings of a recent infrared spectroscopy experiment.<sup>6</sup> Our simulated values of  $D_{H_3O^+}$  show a rapid increase with increasing  $\lambda$  for  $\lambda > 5$ . The presence of multiple  $SO_3^-$  groups around the  $H_3O^+$  ion at low hydration levels impedes the vehicular transport of the  $H_3O^+$  ion. With increasing  $\lambda$ , the average number of  $SO_3^-$  groups around the  $H_3O^+$  ion decreases, <sup>8</sup> which contributes to increasing



**Figure 8.** Index of the  $SO_3^-$  group that is nearest to a  $H_3O^+$  ion for  $\lambda$  values of (a) 7 and (b) 20. An index of 0 indicates a free  $H_3O^+$  ion.

vehicular diffusion of the  $H_3O^+$  ion. Experimental studies have shown that the proton diffusion coefficient<sup>4</sup> is comparable to  $D_{H_3O^{+7}}$  in hydrated Nafion at  $\lambda=3$ , but is an order of magnitude higher at  $\lambda\sim20$ . The first part of this study<sup>8</sup> has shown that the hydration number of  $H_3O^+$  ions increases with increasing  $\lambda$  from  $\sim$ 1.5 at  $\lambda=3$  to  $\sim$ 3.8 at  $\lambda=20$ . This explains the increasing structural diffusion with increasing  $\lambda$ .

Figure 4 shows the MSD of Ow as a function of time for various hydration levels from  $\lambda = 3$  to  $\lambda = 20$ . The data is in good agreement with the results of our earlier simulation of hydrated Nafion at 300 K using a different all-atom force field.9 Figure 5 shows the diffusion coefficient of H<sub>2</sub>O molecules from the present simulation (squares) as a function of hydration level in Nafion.  $D_{\text{H}_2\text{O}}$  increases with increasing  $\lambda$  and attains a value at  $\lambda=20$  that is one-third of the water diffusion coefficient in bulk water.  $^{23}$  The experimental values of  $D_{\rm H_2O}$  obtained from NMR experiments<sup>4</sup> and QENS study<sup>7</sup> of water dynamics in hydrated Nafion are also shown as triangles and circles, respectively. Our simulation results are in good agreement with experimental data over a wide range of hydration levels, which attests to the ability of the all-atom force fields used in this work to model water dynamics in Nafion. In the first part of this work, we showed that the percentage of free (bulklike) water molecules increases from nearly 2.7% at  $\lambda = 3$  to 66% at  $\lambda = 20$ , which is consistent with the increase of  $D_{\rm H_2O}$  with increasing  $\lambda$ . With increasing hydration level, water molecules are able to drift away from the SO<sub>3</sub><sup>-</sup> groups into the water channels, which become increasingly interconnected.8

# 3.2. Mean Residence Time of $H_2O$ Molecules and $H_3O^+$ Ions. We calculated $\tau_{MR}$ of $H_2O$ molecules and $H_3O^+$ ions around each of the $40~SO_3^-$ groups belonging to four Nafion chains at each $\lambda$ . Our simulations revealed the existence of a heterogeneous molecular environment in the simulation cell. Since the algorithm we used for calculating $\tau_{MR}$ is known to give longer residence times<sup>29</sup> than one based on persisting coordination<sup>25–27</sup> and the simulation run was restricted to 2 ns due to computational intensity, some $H_2O$ molecules or $H_3O^+$ ions were found to be bound to $SO_3^-$ groups for the duration of the simulation resulting in $\tau_{MR} \rightarrow \infty$ . This is consistent with the observation in the first part of this work<sup>8</sup> that nearly a third of the water molecules are bound or weakly bound to the $SO_3^-$ groups and $\sim$ 8% of the $H_3O^+$ ions are coordinated by two or more $SO_3^-$ groups at the highest hydration level simulated ( $\lambda$

= 20).

To overcome this difficulty arising from  $H_2O$  molecules or  $H_3O^+$  ions being bound for the duration of the simulation, we determined  $\tau_{MR}$  corresponding to each chain (averaged over  $10~{\rm SO_3}^-$  groups), instead of the whole simulation cell ( $40~{\rm SO_3}^-$  groups), and report here the smallest of the four values of  $\tau_{MR}$  at each  $\lambda$ . Figure 6 shows  $\tau_{MR}$  of  $H_2O$  molecules in one chain of our hydrated Nafion system as a function of  $\lambda$  (squares) along with mean residence time corresponding to slow dynamics ( $\tau_{slow}$ ) observed in QENS experiments (circles). At  $\lambda=3$ ,  $\tau_{MR}$  of  $H_2O$  molecules around a  $SO_3^-$  group is comparable to the total simulation time, which results in a large error bar. With increasing  $\lambda$ ,  $\tau_{MR}$  and  $\tau_{slow}$  decrease and their values are comparable.  $\tau_{MR}$  of  $H_2O$  molecules decreases from  $\sim 1055~{\rm ps}$  at  $\lambda=3$  to  $\sim 75~{\rm ps}$  at  $\lambda=20$ . The experimental value  $^7$  of  $\tau_{slow}$  ranges from 506 ps at  $\lambda=1.95$  to 165 ps at  $\lambda=17.5$ .

We determined  $\tau_{MR}$  of  $H_3O^+$  ions only for  $\lambda = 13.5$  and 20 and these were 572  $\pm$  643 ps and 191  $\pm$  214 ps, respectively. These values are 2.5-4.5 times larger than the corresponding values for  $H_2O$  molecules. At lower  $\lambda$ , each of the chains had at least one bound H<sub>3</sub>O<sup>+</sup> ion for the duration of the simulation, which precluded the determination of  $\tau_{MR}$ . Perrin et al.<sup>7</sup> have offered three possible reasons for the characteristic time of slow protons  $(\tau_{slow})$  observed by QENS, namely, polymer chain dynamics, water clusters trapped in the polymer matrix, and hydronium ion dynamics. The present results are consistent with the third explanation; i.e.,  $\tau_{\text{slow}}$  is associated with the residence of H<sub>3</sub>O<sup>+</sup> ions and H<sub>2</sub>O molecules in the first solvation shell of  $SO_3^-$  groups in Nafion. Our results for  $\tau_{MR}$  of  $H_3O^+$  ions are also consistent with the observation of Perrin et al.<sup>7</sup> that the H<sub>3</sub>O<sup>+</sup> ion lifetime is of the order of or larger than 1 ns and the conclusion of Petersen and Voth12 that sulfonate ions limit hydronium ion diffusion.

3.3. Distance of H<sub>3</sub>O<sup>+</sup> Ions and H<sub>2</sub>O Molecules from the Nearest SO<sub>3</sub><sup>-</sup> Group. The mean residence time discussed in the previous section describes the average behavior of H<sub>3</sub>O<sup>+</sup> ions and H2O molecules. In an effort to shed more light on dynamical behavior in the vicinity of SO<sub>3</sub><sup>-</sup> groups, we have determined the distance from the nearest SO<sub>3</sub><sup>-</sup> group for individual H<sub>3</sub>O<sup>+</sup> ions and H<sub>2</sub>O molecules at 1 ps intervals during our 2 ns simulation. Figure 7a-f shows this distance for a randomly chosen  $H_3O^+$  ion, for  $\lambda = 1, 3, 5, 7, 9$ , and 20, respectively. The distance reported here is between the oxygen of a H<sub>3</sub>O<sup>+</sup> ion and the sulfur of its nearest SO<sub>3</sub><sup>-</sup> group. It is clear from Figure 7a that the H<sub>3</sub>O<sup>+</sup> ion is bound for the duration of the simulation at  $\lambda = 1$  because the data fall entirely below the horizontal line representing the first solvation shell radius (4.3 Å) based on the S-O<sub>h</sub> radial distribution function.<sup>8</sup> At  $\lambda$ = 3 [see Figure 7b], the  $H_3O^+$  ion is bound to the  $SO_3^-$  group

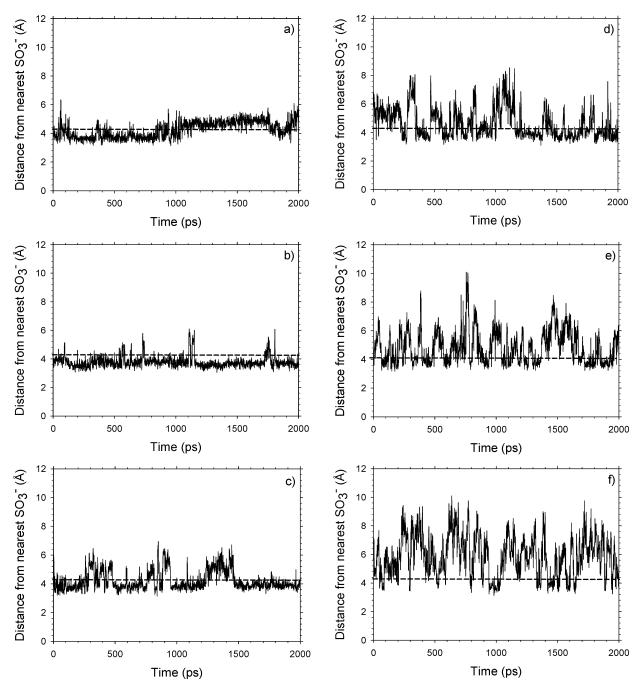
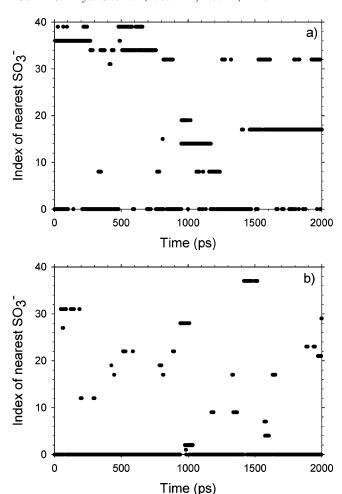


Figure 9. Distance from the nearest SO<sub>3</sub><sup>-</sup> group to a H<sub>2</sub>O molecule for values of (a) 3, (b) 5, (c) 7, (d) 9, (e) 13.5, and (f) 20. The horizontal line represents the radius of the first solvation shell.

but the distance approaches the radius of the first solvation shell. We have observed some H<sub>3</sub>O<sup>+</sup> ions to remain close to the boundary of the solvation shell and transit frequently across this boundary for  $\lambda = 3$ . As  $\lambda$  increases from 5 to 9 [Figure 7c-e)], the average distance of the H<sub>3</sub>O<sup>+</sup> ion from the nearest SO<sub>3</sub><sup>-</sup> group increases. These hydration levels represent a transition from mostly bound (at lower  $\lambda$ ) to mostly free H<sub>3</sub>O<sup>+</sup> ion (at higher  $\lambda$ ). While the variation of the distance to the nearest SO<sub>3</sub><sup>-</sup> group is different for different H<sub>3</sub>O<sup>+</sup> ions, the trend with increasing  $\lambda$  is similar and the average behavior is captured by  $\tau_{\rm MR}$ . At the highest hydration level,  $\lambda = 20$ , shown in Figure 7f, the  $H_3O^+$  ion ventures as far as  $\sim$ 8 Å away from the nearest SO<sub>3</sub><sup>-</sup> group and spends most of the simulation time as a free H<sub>3</sub>O<sup>+</sup> ion. Even at this high hydration level, the H<sub>3</sub>O<sup>+</sup> ion is bound to SO<sub>3</sub><sup>-</sup> groups for a period of the order of 100 ps. Due to frequent transits of H<sub>3</sub>O<sup>+</sup> ions across the solvation shell of

 $SO_3^-$  groups for  $\lambda \geq 5$ , it is possible to get much shorter mean residence times of the order of tens of picoseconds using an algorithm based on persistent coordination.<sup>25–27</sup>

Figure 8a,b shows the index of the nearest SO<sub>3</sub><sup>-</sup> group for the same H<sub>3</sub>O<sup>+</sup> ion discussed above for hydration levels of 7 and 20, respectively. Indices of 1-10, 11-20, 21-30, and 31-40 represent the four chains, while an index of 0 indicates that no sulfur atom is present within 4.3 Å of the  $O_h$  (free  $H_3O^+$ ). There is at most one SO<sub>3</sub><sup>-</sup> nearest neighbor at a given time step, but the figure gives the appearance of multiple indices at a given time, because we have plotted 2000 data points that often lie close together. The H<sub>3</sub>O<sup>+</sup> ion is coordinated by different SO<sub>3</sub><sup>-</sup> groups belonging to different chains at various times during the course of the simulation, which indicates the H<sub>3</sub>O<sup>+</sup> ion is able to move from one SO<sub>3</sub><sup>-</sup> group to another even though its mean residence time is of the order of 1 ns. When viewed



**Figure 10.** Index of the  $SO_3^-$  group that is nearest to a  $H_2O$  molecule for  $\lambda$  values of (a) 7 and (b) 20. An index of 0 indicates a free  $H_2O$  molecule.

together, Figures 7 and 8 provide valuable information about the path traveled by the  $H_3O^+$  ion. The  $H_3O^+$  ion spends some time bound to  $SO_3^-$  groups, moves away from the  $SO_3^-$  group toward the center of the Nafion nanopore, and subsequently becomes bound to a  $SO_3^-$  group that may be different from the one previously binding the  $H_3O^+$  ion. The  $H_3O^+$  ion residence time decreases and the frequency of the transit between  $SO_3^-$  groups increases as the hydration level increases. At low hydration level (say  $\lambda=7$ ), it is possible for the  $H_3O^+$  ion to change its binding from one  $SO_3^-$  group to another without becoming free because there are several  $SO_3^-$  groups close to the  $H_3O^+$  ion.

Figure 9a-f shows the distance between the oxygen of a randomly chosen H<sub>2</sub>O molecule and the sulfur of the nearest  $SO_3^-$  group for  $\lambda = 3, 5, 7, 9, 13.5$ , and 20, respectively. The trends in the data are similar to that seen in Figure 7 for H<sub>3</sub>O<sup>+</sup> ions. However, at a given  $\lambda$ , the H<sub>2</sub>O molecule remains bound to the nearest SO<sub>3</sub><sup>-</sup> group for a shorter time and travels farther from the SO<sub>3</sub><sup>-</sup> group compared to H<sub>3</sub>O<sup>+</sup> ions. At all hydration levels examined, the H<sub>2</sub>O molecule spends some time being bound to the SO<sub>3</sub><sup>-</sup> group, but this time decreases with increasing  $\lambda$ . Figure 10, parts (a) and (b), shows the index of the SO<sub>3</sub> group nearest to the H<sub>2</sub>O molecule represented in Figure 9, parts (c) and (f), corresponding to hydration levels of 7 and 20, respectively. As in the case of the H<sub>3</sub>O<sup>+</sup> ion, the H<sub>2</sub>O molecule is coordinated by different SO<sub>3</sub><sup>-</sup> groups during the simulation. At  $\lambda = 20$ , the H<sub>2</sub>O molecule is mostly free (index = 0), but binds for periods of the order of tens of picoseconds to SO<sub>3</sub><sup>-</sup>

groups belonging to all four chains. At the highest hydration level examined [see Figure 9f], the  $H_2O$  molecule ventures  $\sim 10$  Å from the nearest  $SO_3^-$  group.

We have found 11 Å to be the maximum  $S-O_h$  separation after examining the data for randomly chosen  $H_2O$  molecules at each  $\lambda$ . Gebel<sup>30</sup> has proposed a model of hydrated Nafion based on small-angle X-ray scattering that considers the dry membrane to be made of isolated spherical ionic clusters of radius  $\sim 7.5$  Å that swell with increasing hydration. In this model, the hydrated membrane consists of pools of water of radius  $\sim 10$  Å surrounded by ionic groups at the polymer—water interface.<sup>30,31</sup> At high levels of hydration, water channels connect the spherical clusters. Our finding of a maximum separation of  $\sim 11$  Å between a typical  $H_2O$  molecule and its nearest  $SO_3$ -neighbor at  $\lambda = 20$  is consistent with Gebel's model<sup>30</sup> of a spherical pool of water of radius 10 Å in hydrated Nafion.

The findings of this two-part work are as follows. At low hydration levels ( $\lambda \leq 5$ ) in Nafion,  $H_3O^+$  ions are coordinated by multiple  $SO_3^-$  groups and fewer than 20% of the  $H_2O$  molecules are free (bulklike). More than 50% of the hydronium ions exist in bridging configurations between sulfonate groups. As the hydration level increases, the sulfonate groups move apart and the hydration number of hydronium ions increases. The water molecules mediate the interaction between hydronium ions and sulfonate groups, leading to an increase in their separation distance. With increasing hydration level, the diffusion coefficients of  $H_3O^+$  ions and  $H_2O$  molecules increase and the mean residence time of water molecules around sulfonate groups decreases.

The calculation of mean residence times of  $H_2O$  molecules and  $H_3O^+$  ions in Nafion at various hydration levels by a rigorous method that avoids difficulties arising from frequent transit across the solvation shell boundary is an important contribution of this work. The long residence times ( $\sim 1$  ns) of  $H_2O$  molecules and  $H_3O^+$  ions indicates that longer scale simulations may be needed to understand dynamical behavior at low hydration levels. Our results provide a molecular-level explanation for the conclusion of Kreuer et al.  $^{10}$  and Zawodzinski et al.  $^4$  that vehicular proton transport is dominant at low hydration levels while structural diffusion dominates proton transport at high hydration levels in Nafion.

## 4. Conclusions

We have calculated the diffusion coefficients of water molecules in Nafion at various hydration levels and found them to be in good agreement with experimental data. At high hydration levels, the calculated diffusion coefficient of hydronium ions is in agreement with the value reported by a recent QENS experiment. Our simulations reveal that H<sub>2</sub>O molecules and H<sub>3</sub>O<sup>+</sup> ions migrate by diffusion between the solvation shell of different SO<sub>3</sub><sup>-</sup> groups and the center of the Nafion pore. The mean residence time of H<sub>2</sub>O molecules near sulfonate groups decreases with increasing hydration level from ~1 ns at  $\lambda = 5$  to  $\sim 75$  ps at  $\lambda = 20$ . The mean residence time of H<sub>3</sub>O<sup>+</sup> ions is larger than the corresponding values for H<sub>2</sub>O molecules by a factor of 2.5-4.5. Our work provides an explanation for the experimentally observed characteristic time of slow proton dynamics in hydrated Nafion in terms of the residence of H<sub>3</sub>O<sup>+</sup> ions and H<sub>2</sub>O molecules in the first solvation shell of SO<sub>3</sub><sup>-</sup> groups. The present results in conjunction with the results from the first part of this work establish an important link between Nafion membrane nanostructure and the dynamics of H<sub>2</sub>O molecules and H<sub>3</sub>O<sup>+</sup> ions.

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### **References and Notes**

- (1) Crabtree, G. W.; Dresselhaus, M. S.; Buchanan, M. V. *Phys. Today* **2004**, *57* (12), 39.
- (2) Eikerling, M.; Kornyshev, A. A.; Kucernak, A. R. *Phys. Today* **2006**, *59* (10), 38.
- (3) Paciaroni, A.; Casciola, M.; Cornicchi, E.; Marconi, M.; Onori, G.; Pica, M.; Narducci, R. J. Phys. Chem. B 2006, 110, 13769.
- (4) Zawodzinski, T. A.; Neeman, M.; Sillerud, L. O.; Gottesfeld, S. J. Phys. Chem. **1991**, 95, 6040.
  - (5) Pivovar, A. M.; Pivovar, B. S. J. Phys. Chem. B **2005**, 109, 785.
- (6) Moilanen, D. E.; Piletic, I. R.; Fayer, M. D. J. Phys. Chem. A 2006, 110, 9084.
- (7) Perrin, J.-C.; Lyonnard, S.; Volino, F. J. Phys. Chem. C 2007, 111, 3393.
- (8) Devanathan, R.; Venkatnathan, A.; Dupuis, M. J. Phys. Chem. B 2007, 111, 8069.
- (9) Venkatnathan, A.; Devanathan, R.; Dupuis, M. J. Phys. Chem. B 2007, 111, 7234.
- (10) Kreuer, K.-D.; Paddison, S. J.; Spohr, E.; Schuster, M. Chem. Rev. 2004, 104, 4637.

- (11) Marx, D.; Tuckerman, M. E.; Hutter, J.; Parrinello, M. Nature 1999, 397, 601.
  - (12) Petersen, M. K.; Voth, G. A. J. Phys. Chem. B 2006, 110, 18594.
  - (13) Paul, R.; Paddison, S. J. Chem. Phys. 2005, 123, 224704.
- (14) Jang, S. S.; Molinero, V.; Çağin, T.; Goddard, W. A. J. Phys. Chem. B 2004, 108, 3149.
- (15) Urata, S.; Irisawa, J.; Takada, A.; Shinoda, W.; Tsuzuki, S.; Mikami, M. J. Phys. Chem. B 2005, 109, 4269.
- (16) Cui, S.; Liu, J.; Selvan, M. E.; Keffer, D. J.; Edwards, B. J.; Steele, W. V. J. Phys. Chem. B **2007**, 111, 2208.
- (17) Blake, N. P.; Mills, G.; Metiu, H. J. Phys. Chem. B 2007, 111,
- (18) Elliott, J. A.; Paddison, S. J. Phys. Chem. Chem. Phys. 2007, 9, 2602.
- (19) Schröder, C.; Rudas, T.; Boresch, S.; Steinhauser, O. J. Chem. Phys. **2006**, 124, 234907.
  - (20) Paddison, S. J.; Elliott, J. A. J. Phys. Chem. A 2005, 109, 7583.
  - (21) Smith, W.; Forester, T. R. J. Mol. Graphics 1996, 14, 136.
- (22) Mayo, S. L.; Olafson, B. D.; Goddard, W. A. J. Phys. Chem. 1990, 94, 8897.
- (23) Levitt, M.; Hirshberg, M.; Sharon, R.; Laidig, K. E.; Daggett, V. J. Phys. Chem. B **1997**, 101, 5051.
  - (24) Morris, D. R.; Sun, X. J. Appl. Polym. Sci. 1993, 50, 1445.
- (25) Impey, R. W.; Madden, P. A.; McDonald, I. R. J. Phys. Chem. 1983, 87, 5071.
- (26) Luise, A.; Falconi, M.; Desideri, A. Proteins: Struct. Funct. Genet. 2000, 39, 56.
  - (27) Garcia, A. E.; Stiller, L. J. Comput. Chem. 1993, 14, 1396.
- (28) Brunne, R. M.; Liepinsh, E.; Otting, G.; Wüthrich, K.; van Gunsteren, W. F. *J. Mol. Biol.* **1993**, *231*, 1040.
- (29) Schoenborn, B. P.; Garcia, A.; Knott, R. Prog. Biophys. Mol. Biol. 1995, 64, 105.
  - (30) Gebel, G. Polymer 2000, 41, 5829.
  - (31) Mauritz, K. A.; Moore, R. B. Chem. Rev. 2004, 104, 4535.