

# Mesoscale Multi-Physics Simulation of Solidification in Selective Laser Melting Process Using A Phase Field and Thermal Lattice Boltzmann Model

Dehao Liu, Yan Wang\*

Georgia Institute of Technology

[yan.wang@me.gatech.edu](mailto:yan.wang@me.gatech.edu)

<http://msse.gatech.edu>

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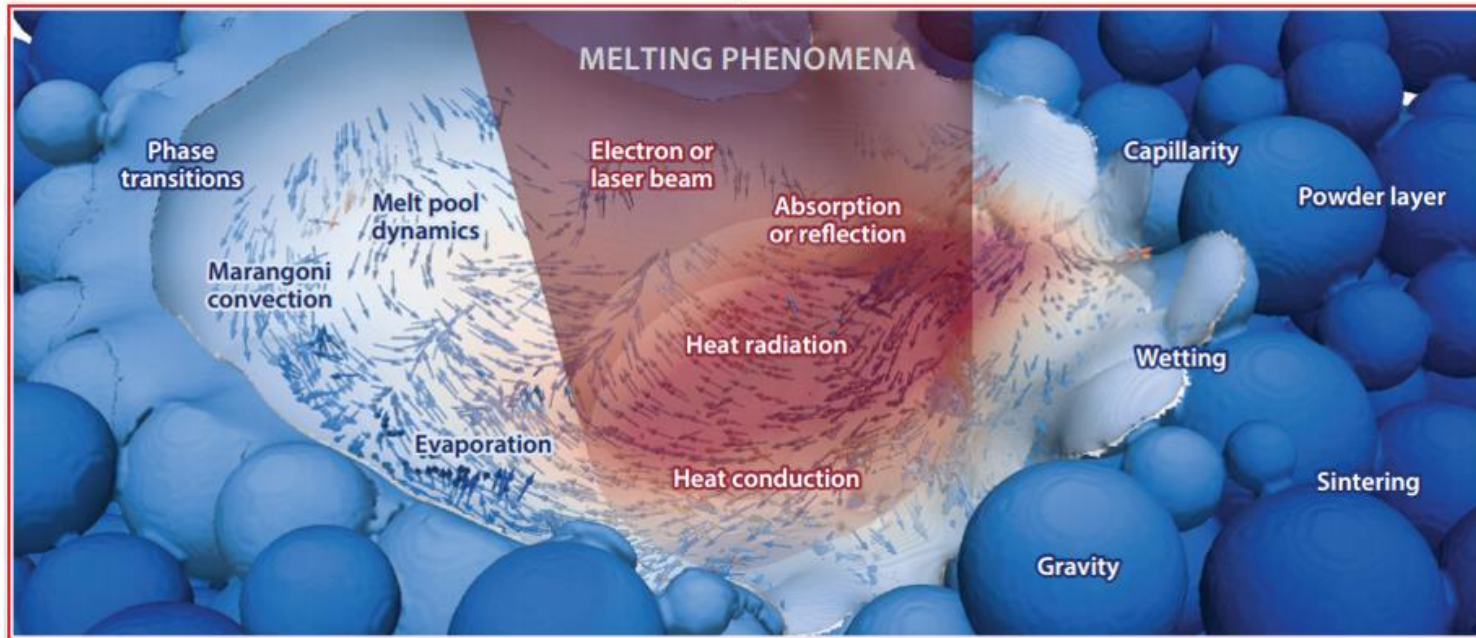
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# Outline

- **Background**
- **Methodology**
  - Phase Field Method
  - Thermal Lattice Boltzmann
  - Motion of Grain
  - Simulation Algorithm
- **Simulation Results**
  - Effect of Flow
  - Effect of Cooling Rate
- **Summary**

# Background

- Solidification of melt in SLM process is very complicated, which involves multiple physical phenomena
- Challenge: Create a multi-physics based model to investigate the Process-Structure relationship



Markl, M., & Körner, C. (2016)

# Single-Physics Simulation

## – Phase Field Method

- Phase field method has been used to simulate the microstructure evolution and solute concentration of Ti-6Al-4V alloy during solidification in powder-bed electron beam AM process (Gong & Chou 2015; Sahoo & Chou 2016)

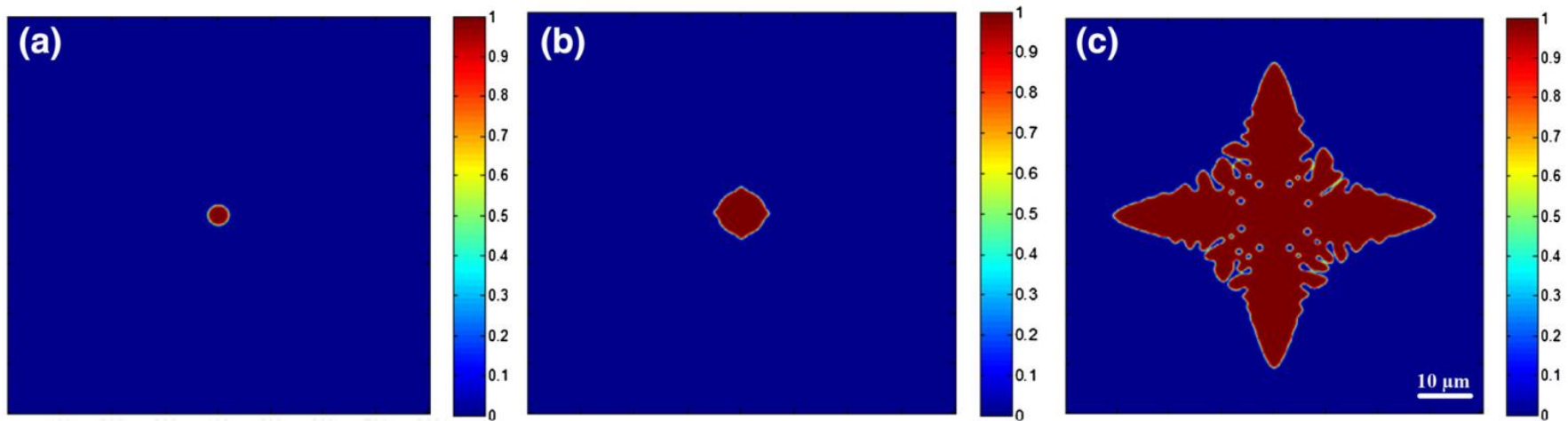


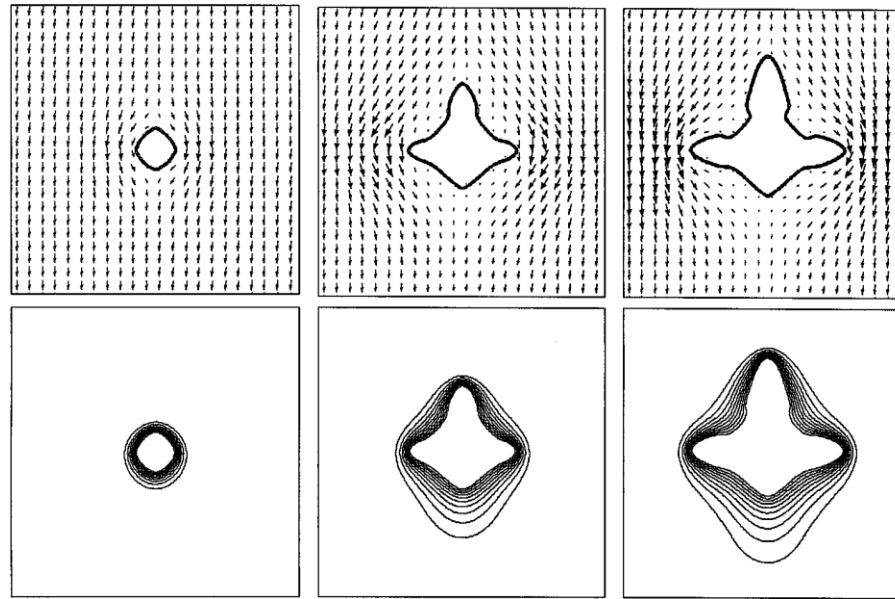
Fig. 1. Simulated dendrite structure growth at different times: (a) 0.01 ms, (b) 0.2 ms, and (c) 1.6 ms.

- Isothermal assumption without considering the effect of the latent heat
- No effect of melt flow

# Multi-Physics Simulation

## – Phase Field + Convection

- Navier-Stokes equation to predict fluid flow velocity (Beckermann et al. 1999)
- Advection-Diffusion

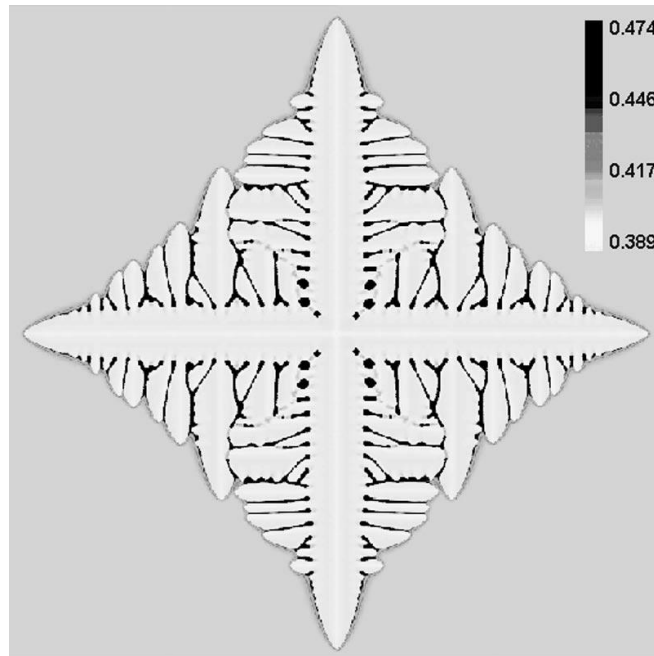


(Beckermann et al. 1999)

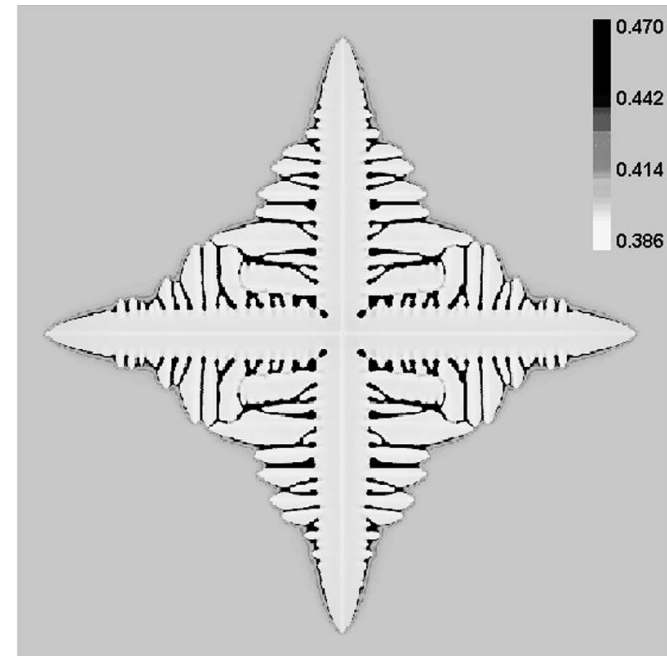
# Multi-Physics Simulation

## – Phase Field + Non-isothermal

- Heat transfer with high cooling rates (Loginova et al. 2001; Grujicic et al. 2002; Echebarria et al. 2004)



Isothermal



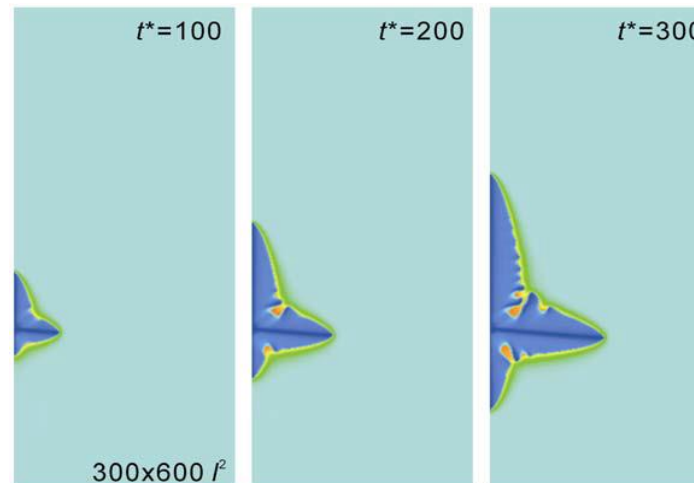
Non-isothermal

(Loginova et al., 2001)

# Multi-Physics Simulation

– Phase Field + Convection + Non-isothermal

- Navier-Stokes equation is applied to predict fluid flow velocity, thermal conduction (Lan & Shih, 2004; Du & Zhang, 2014; Holfelder et al. 2016)
- Advection-diffusion



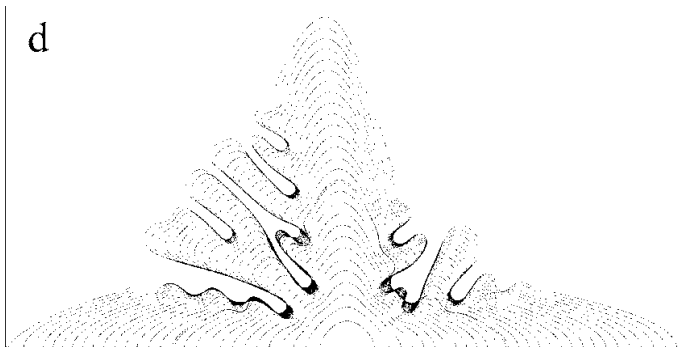
(Lan & Shih, 2004)



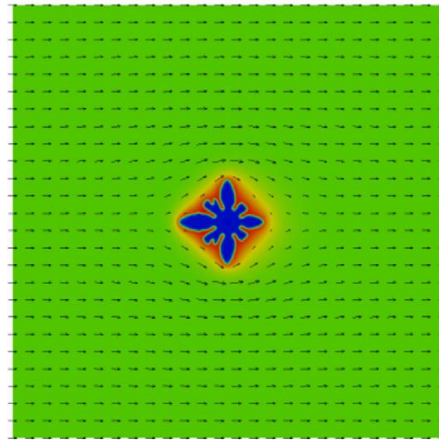
# Multi-Physics Simulation

– Phase field + lattice Boltzmann method

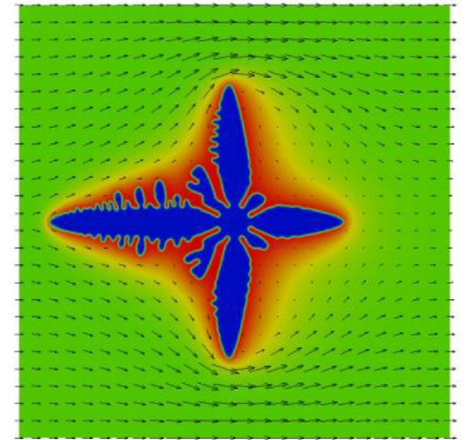
- Lattice Boltzmann method simulates fluid flow (Medvedev & Kassner 2005; Miller et al. 2006; ; Rojas et al. 2015; Böttger et al. 2016; Cartalade et al. 2016)



(Medvedev & Kassner 2005)



(Rojas et al. 2015)

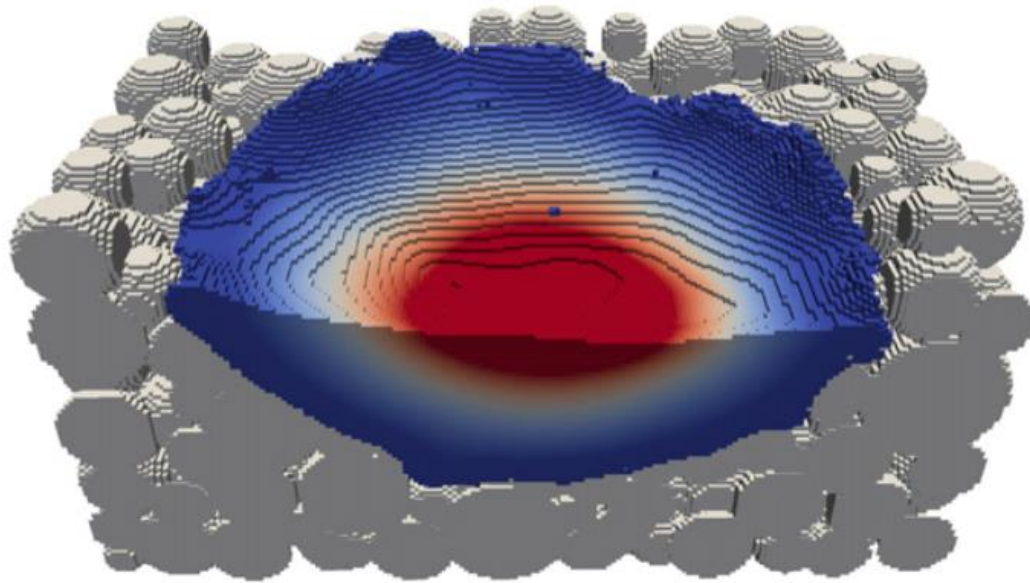




# Multi-Physics Simulation

## – thermal lattice Boltzmann method

- 3D thermal lattice Boltzmann method to simulate the evolution of temperature and velocity field in electron beam melting processes (Ammer et al., 2014)



(Ammer et al., 2014)

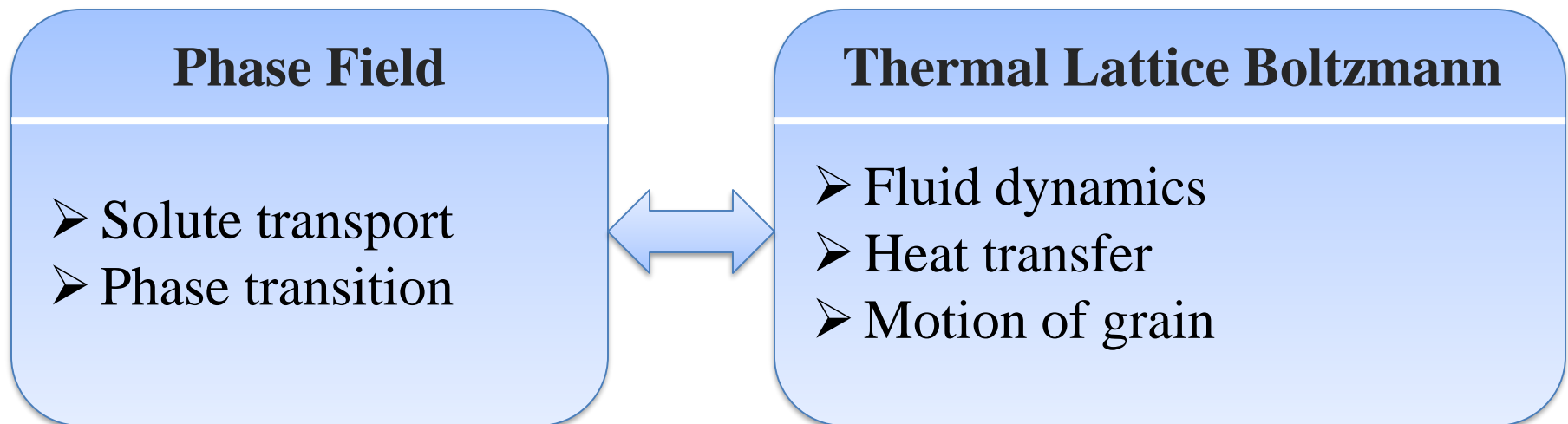
- The evolution of dendrite structure is not simulated

- **Methodology**

- Phase Field Method
- Thermal Lattice Boltzmann
- Motion of Grain
- Simulation Algorithm

# Proposed Multi-Physics Model

- Phase Field + Thermal Lattice Boltzmann Method (**PF-TLBM**) integrates:
  - solute transport,
  - heat transfer,
  - phase transition,
  - fluid dynamics,
  - motion of the solid phase



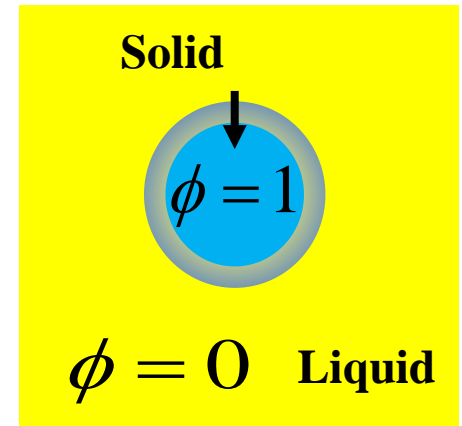
# Phase Field Method

- Phase Field Method (PFM) is a versatile and accurate numerical tool to simulate solidification (Boettinger et al., 2002; Chen, 2002; Singer-Loginova and Singer, 2008; Moelans et al., 2008; Steinbach, 2009)
- *Phase field* or *order parameter*  $\phi$  describes the distribution of phase
- Free energy functional drives the evolution of microstructure

$$F = \int_{\Omega} (f^{GB} + f^{CH}) dV$$

$$f^{GB} = \frac{4\sigma(\mathbf{n})}{\eta} \left\{ |\nabla \phi|^2 + \frac{\pi^2}{\eta^2} \phi(1-\phi) \right\}$$

$$f^{CH} = h(\phi) f_s(C_s) + h(1-\phi) f_l(C_l) + \mu(C - (\phi_s C_s + \phi_l C_l))$$



# Our Implemented PFM

- Kinetic equation for the phase field

$$\dot{\phi} + \boxed{\mathbf{u}_s \cdot \nabla \phi} = M_\phi \left\{ \sigma(\mathbf{n}) \left[ \nabla^2 \phi + \frac{\pi^2}{\eta^2} \left( \phi - \frac{1}{2} \right) \right] + \frac{\pi}{\eta} \sqrt{\phi(1-\phi)} \Delta G \right\}$$

- Kinetic equation for the composition field

$$\dot{C} + \boxed{\mathbf{u}_l \cdot \nabla ((1-\phi)C_l) + \mathbf{u}_s \cdot \nabla (\phi C_s)} = \nabla \cdot (D_l (1-\phi) \nabla C_l) + \boxed{\nabla \cdot \mathbf{j}_{at}}$$

- Advection effect is considered in solute transport and phase transition
- Anti-trapping current

$$\mathbf{j}_{at} = \frac{\eta}{\pi} \sqrt{\phi(1-\phi)} (C_l - C_s) \dot{\phi} \frac{\nabla \phi}{|\nabla \phi|}$$

# Thermal Lattice Boltzmann Method

- Coupling of melt flow with heat transfer

$$\begin{aligned}\nabla \cdot (\phi_l \mathbf{u}_l) &= 0 \\ \frac{\partial}{\partial t} (\phi_l \mathbf{u}_l) + \nabla \cdot (\phi_l \mathbf{u}_l \mathbf{u}_l) &= -\frac{\phi_l}{\rho} \nabla P + \nabla \cdot [\nu \nabla (\phi_l \mathbf{u}_l)] + \mathbf{F} \\ \frac{\partial T}{\partial t} + \nabla \cdot (\mathbf{u}_l T) &= \nabla \cdot (\alpha \nabla T) + q\end{aligned}$$

- Kinetic equations for *density* and *temperature* particle distributions

*Density:* 
$$f_i(\mathbf{x} + \mathbf{e}_i \Delta t, t + \Delta t) - f_i(\mathbf{x}, t) = \frac{1}{\tau_f} (f_i^{eq}(\mathbf{x}, t) - f_i(\mathbf{x}, t)) + F_i(\mathbf{x}, t)$$

*Temperature:* 
$$g_i(\mathbf{x} + \mathbf{e}_i \Delta t, t + \Delta t) - g_i(\mathbf{x}, t) = \frac{1}{\tau_g} (g_i^{eq}(\mathbf{x}, t) - g_i(\mathbf{x}, t)) + Q_i(\mathbf{x}, t)$$

- Macroscopic quantities of density, velocity, and temperature are calculated from  $f_i$ 's and  $g_i$ 's.



# Thermal Lattice Boltzmann Method

- Latent heat

$$q = \frac{L_H}{c_p} \frac{\partial \phi}{\partial t}$$

- Force source

$$F_i = \left(1 - \frac{1}{2\tau_f}\right) \omega_i \left( \frac{\mathbf{e}_i - \mathbf{u}_l}{c_s^2} + \frac{\mathbf{e}_i \cdot \mathbf{u}_l}{c_s^4} \mathbf{e}_i \right) \cdot \mathbf{F}$$

- Heat source

$$Q_i = \left(1 - \frac{1}{2\tau_g}\right) \omega_i q$$

- weights (for D2Q9)

$$\omega_i = \begin{cases} 4/9, & i = 0 \\ 1/9, & i = 1, \dots, 4 \\ 1/36, & i = 5, \dots, 8 \end{cases}$$

# Motion of Grain

- Motion of grain includes rigid translation and rotation of the grains.

- Total force and torque acting on a grain

$$\mathbf{F} = -\sum \mathbf{F}_d, \quad \mathbf{M} = -\sum (\mathbf{r} - \mathbf{R}_{cm}) \times \mathbf{F}_d$$

- Translation of the grain (center of mass)

$$\dot{\mathbf{R}}_{cm} = \mathbf{U}_{cm}, \quad \dot{\mathbf{U}}_{cm} = \mathbf{F} / m$$

- Rotation of the grain

$$\theta = \omega, \quad \dot{\omega} = \mathbf{M} / \mathbf{I}$$

- Local velocity of the grain

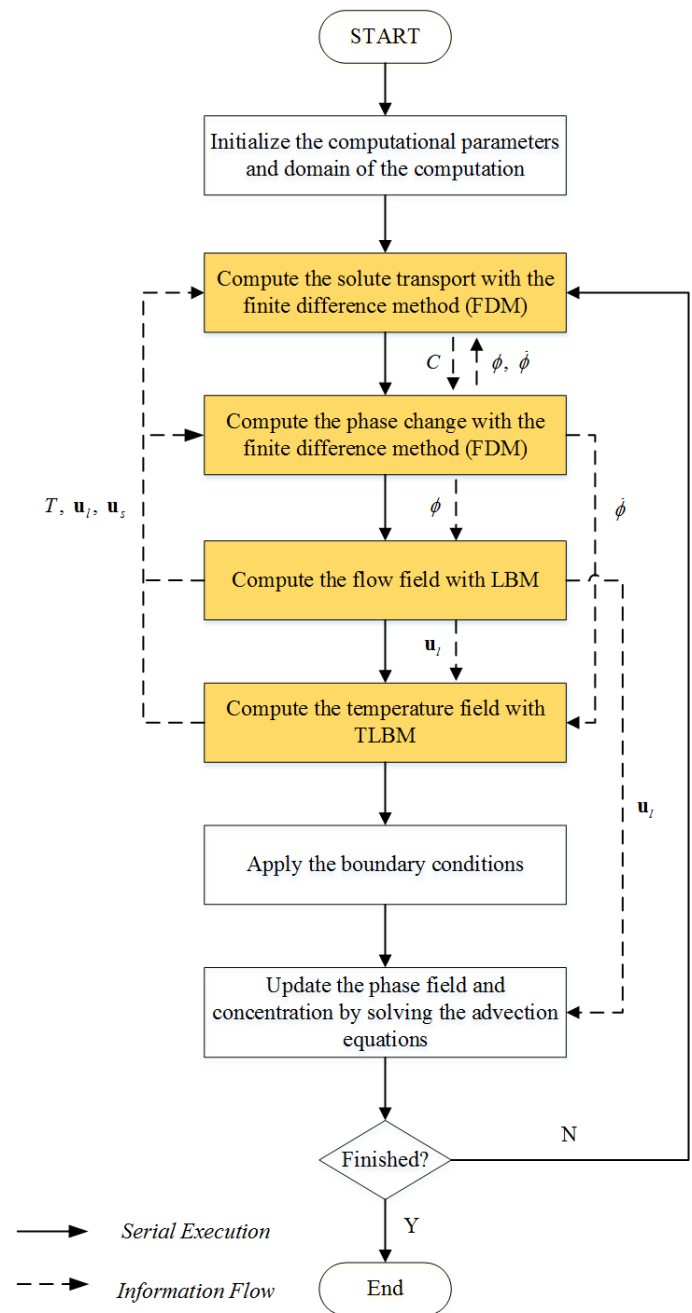
$$\mathbf{u}_s = \mathbf{U}_{cm} + \omega \times (\mathbf{r} - \mathbf{R}_{cm})$$

# PF-TLBM Algorithm

- Different variables are coupled

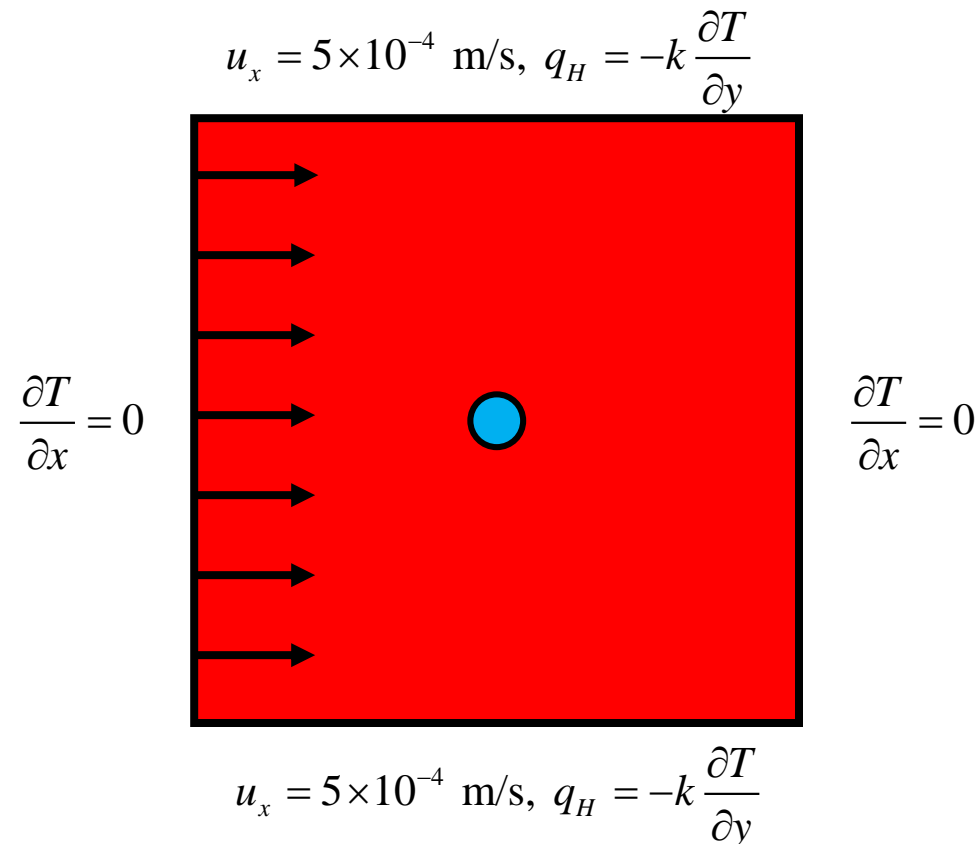
$$\phi \quad \dot{\phi} \quad C \quad T \quad \mathbf{u}_l \quad \mathbf{u}_s$$

- The algorithm is implemented in C++ programming language and integrated with OpenPhase
- The OpenMP shared-memory parallel programming framework is used to accelerate the computation



# Simulation Results: Al-Cu

- A single nucleus is put at the center of Al-4wt%Cu alloy melt flow
- Zero Neumann conditions are applied at all boundaries for phase field  $\phi$  and composition  $C$ .
- For melt flow, periodic boundary conditions are applied at the left and right boundary
- Given a constant cooling rate  $\dot{T} = 2 \times 10^4$  K/s, a fixed heat flux  $q_H = \rho c_p \dot{T} L_y / 2$  is set at the upper and lower boundaries



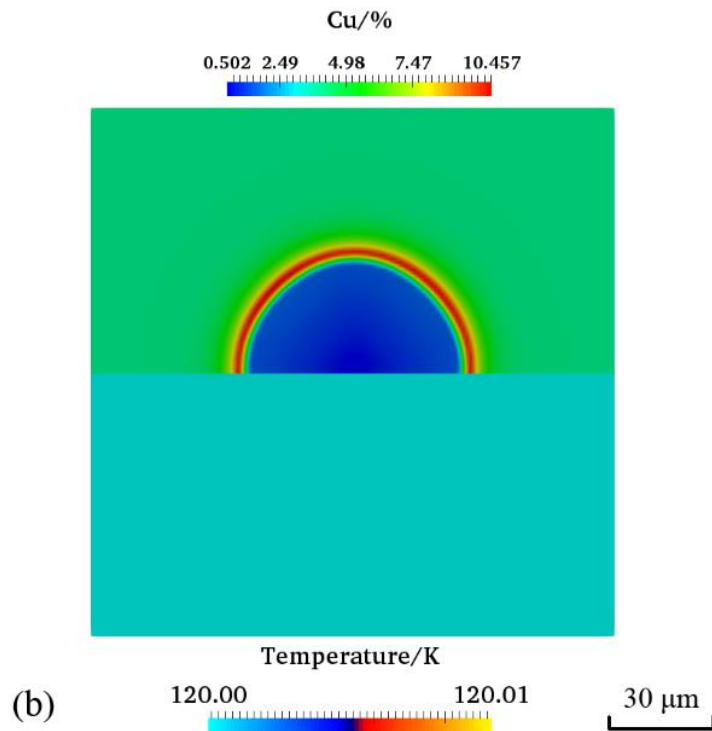
# Physical properties

**TABLE 1:** Physical properties of Al-4wt%Cu alloy

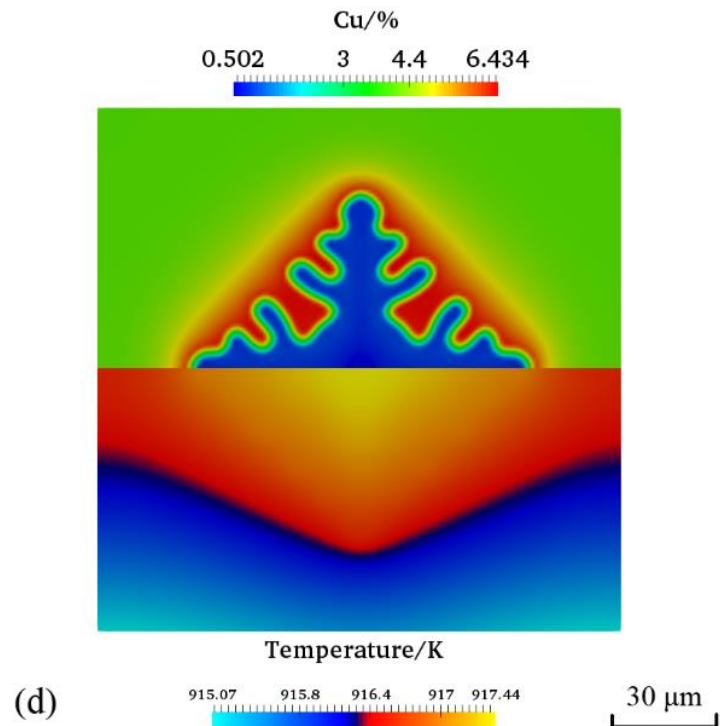
Melting point Al	$T_m$ [K]	933.6
Liquidus slope	$m_l$ [K/%]	-2.6
Partition coefficient	$k$	0.14
Liquid diffusivity	$D_l$ [m <sup>2</sup> /s]	$3.0 \times 10^{-9}$
Interface energy	$\sigma_0$ [J/ m <sup>2</sup> ]	0.24
Interface energy anisotropy	$\varepsilon$	0.35
Kinematic viscosity	$\nu$ [m <sup>2</sup> /s]	$5.7 \times 10^{-7}$
Thermal diffusivity	$\alpha$ [m <sup>2</sup> /s]	$4.9 \times 10^{-5}$
Latent heat of fusion	$L_H$ [ J/kg ]	$3.98 \times 10^5$
Specific heat capacity	$c_p$ [ J/(kg•K) ]	1450

# Simulation Results: Al-Cu

- isothermal



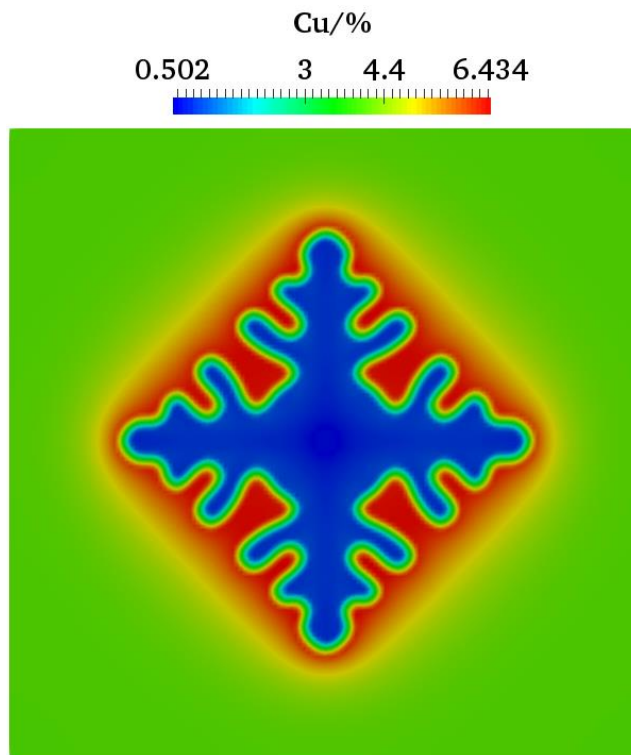
- Non-isothermal



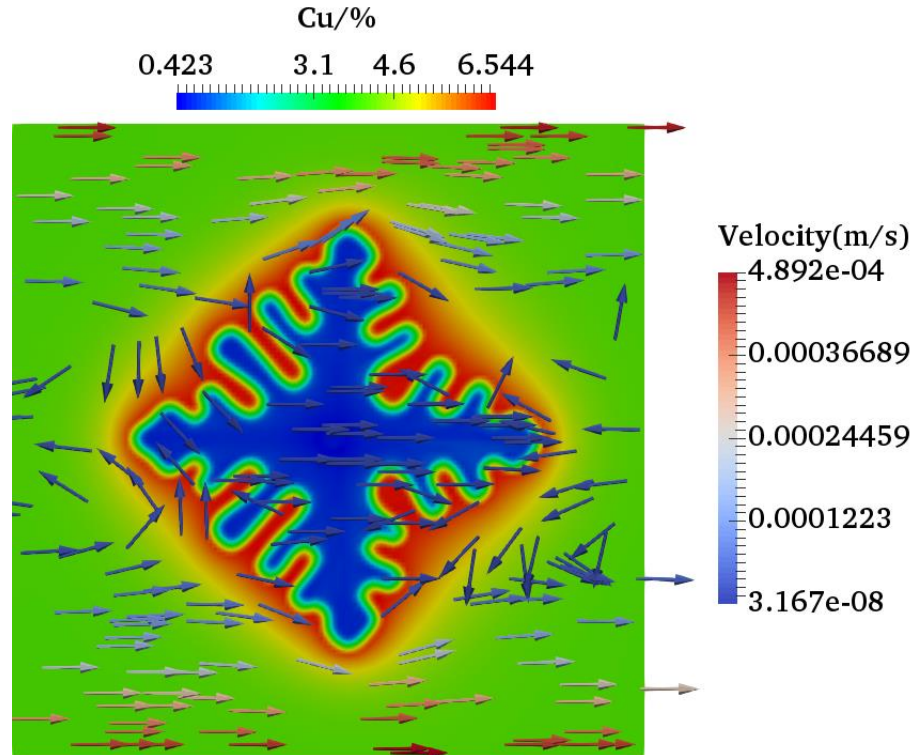


# Effect of Flow

- With flow, the upstream portion of dendrite grows faster than the downstream portion, and the primary arm against the flow is deflected



100ms

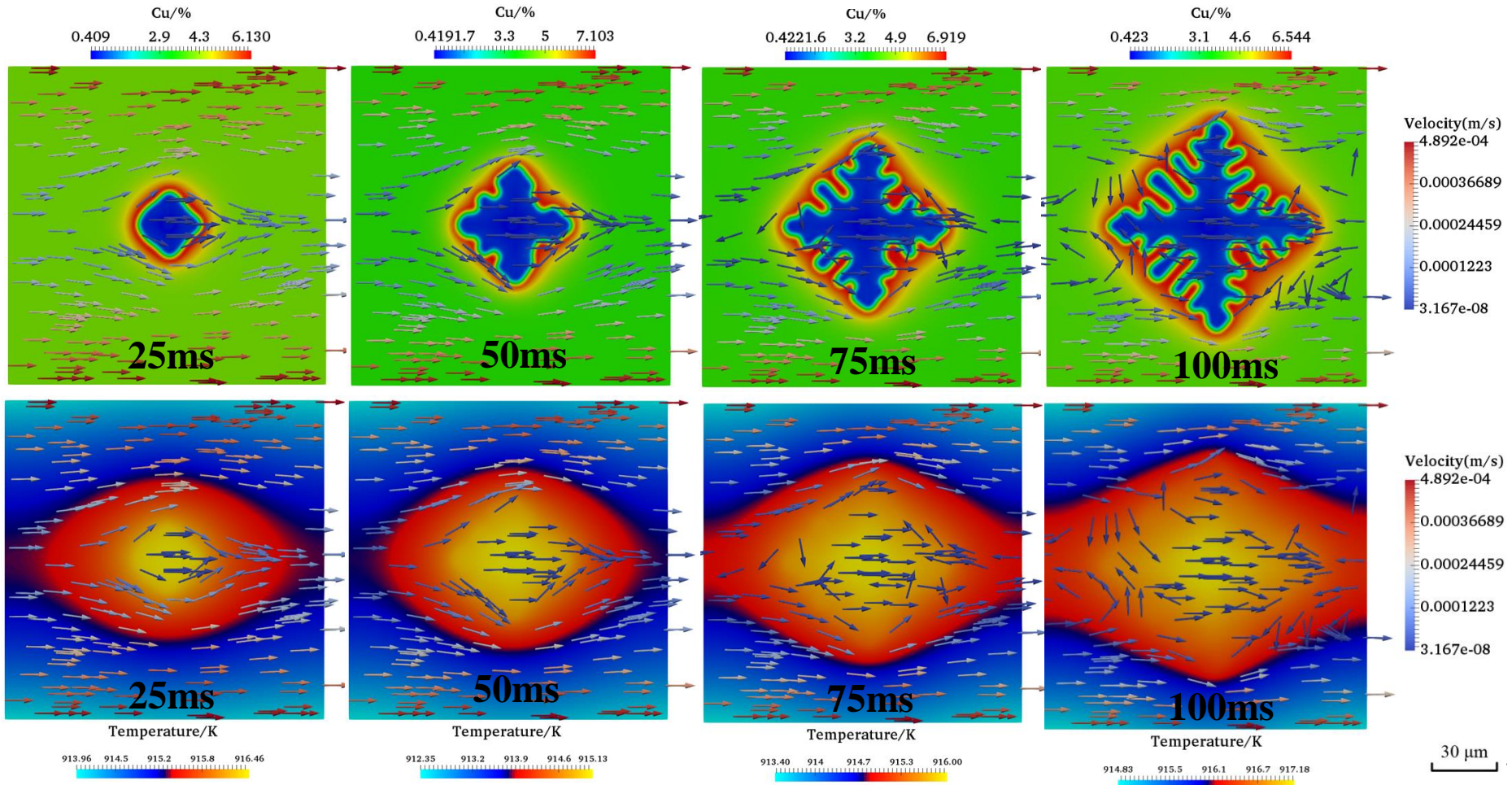


100ms

30 μm

# Effect of Flow

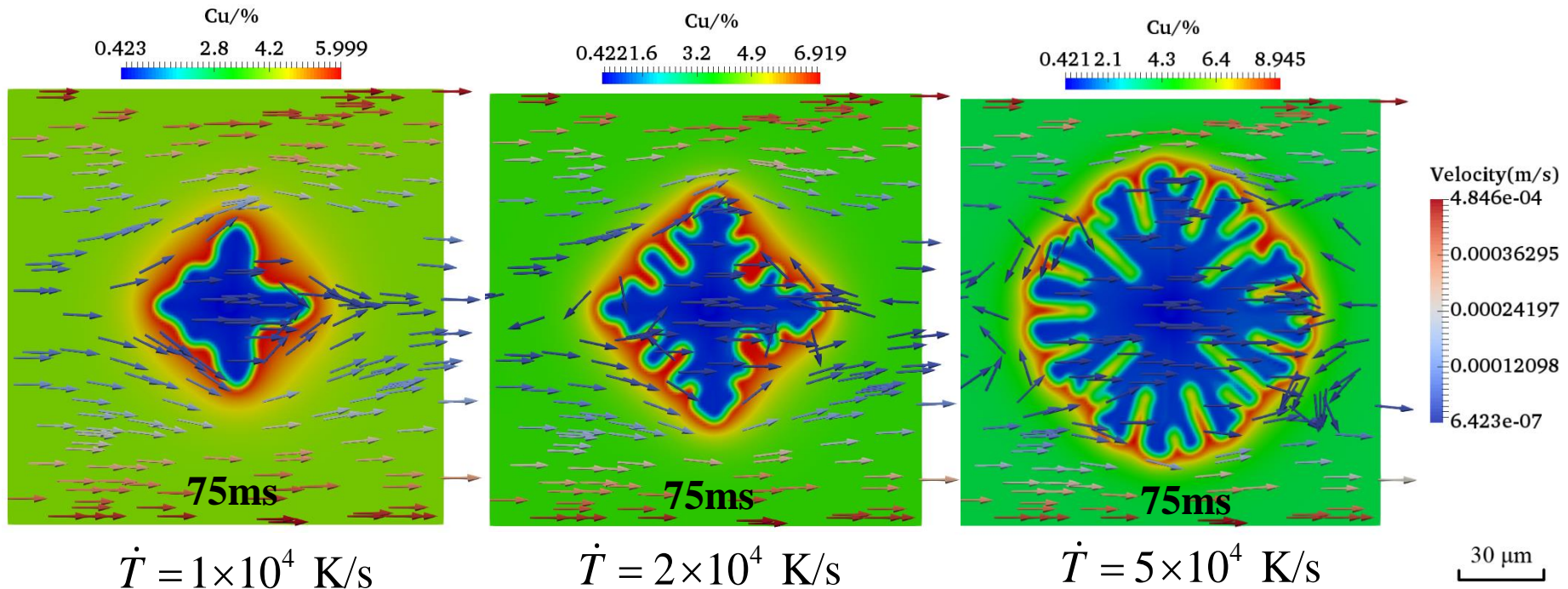
- The flow brings fresh material to its vicinity, which also increases the undercooling



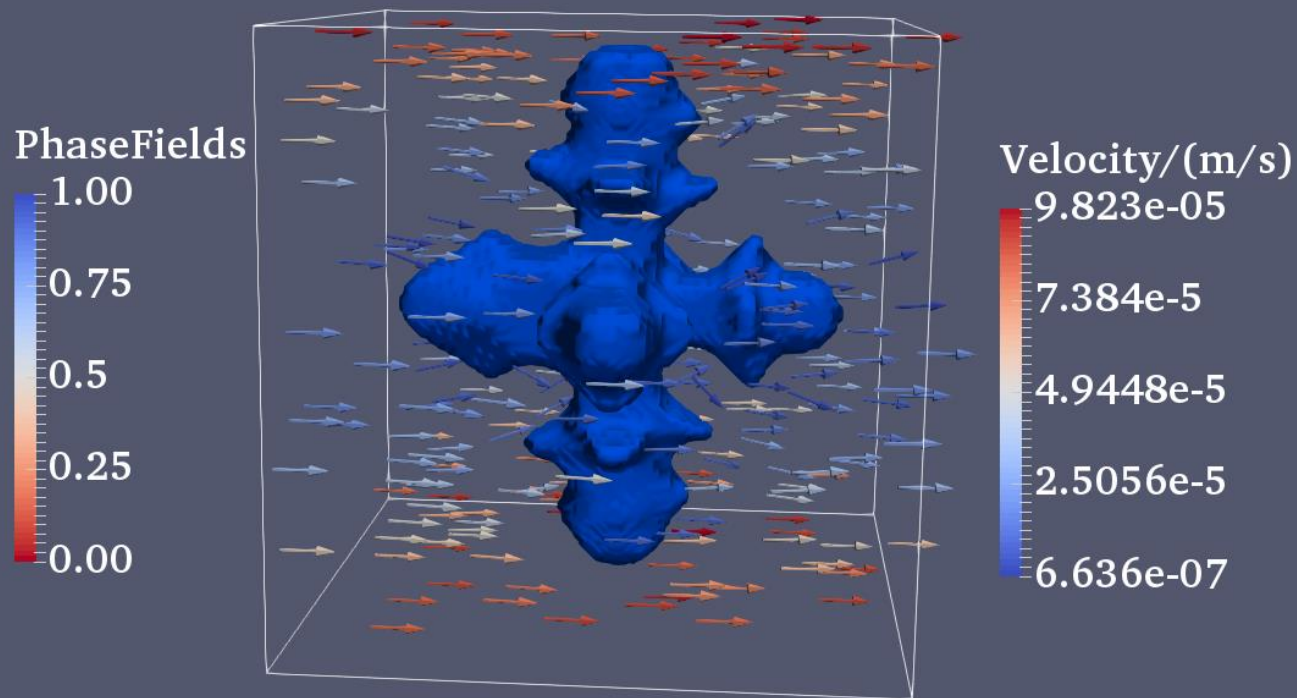


# Effect of Cooling Rate

- Higher cooling rate increases the growth rate of secondary arms of dendrite
- A higher cooling rate also results in higher segregation of Cu at the solid-liquid interface

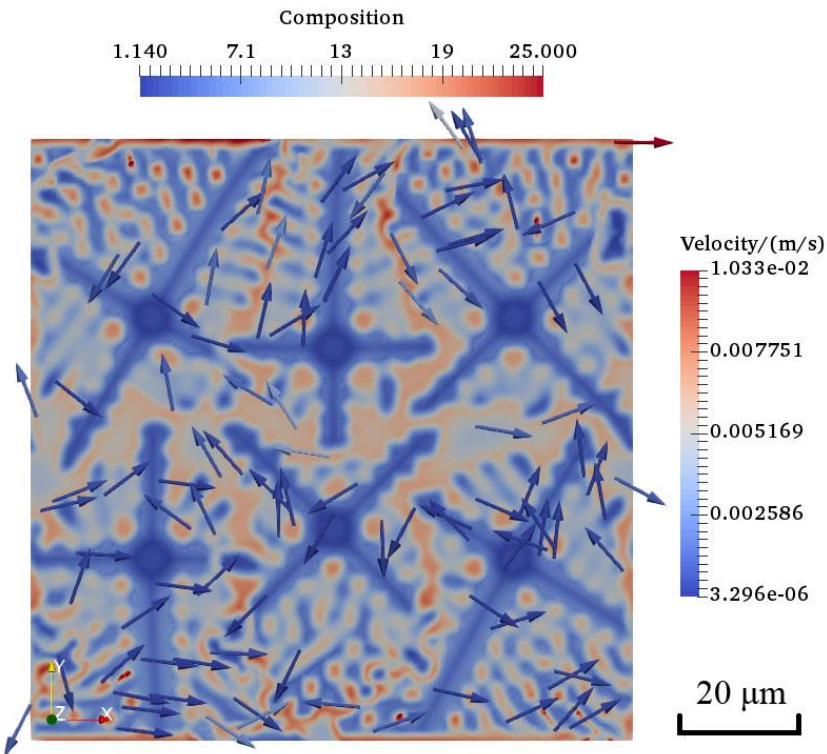


# 3D Dendrite Growth

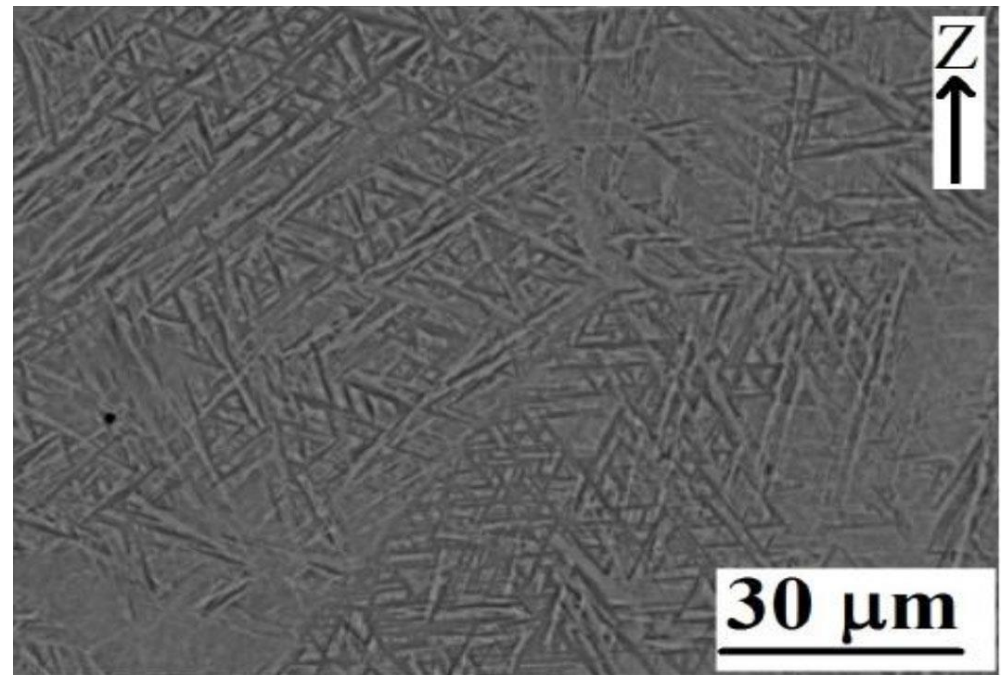


# Simulation Results: Ti-6Al-4V

- Multigrain growth
- Cooling rate:  $\dot{T} = 2 \times 10^5$  K/s
- Flow speed: 50 mm/s



**Simulated microstructure**



**SEM image of Ti64 microstructure produced by SLM**

# Research Issues

- How to improve computational efficiency
  - parallelization
- How to improve accuracy
  - Uncertainty quantification
    - Parameter uncertainty
    - Model form uncertainty
- How to use simulation in process planning and optimization
  - Establishing Process-Structure-Property (P-S-P) relationship



# Summary

- A mesoscale multi-physics model is developed to simulate the solidification process in SLM based on Phase Field and Thermal Lattice Boltzmann Methods
- The PF-TLBM model incorporates solute transport, heat transfer, fluid dynamics, kinetics of phase transformation, and grain growth.
- It simulates systems at a reasonable time scale for manufacturing processes while providing fine-grained material phase and composition information.

*köszönöm ! תודה dēkuji*

*mahalo* 고맙습니다

*thank you*

**Thanks!**

*merci* 谢谢 *danke*

*Eucharistó* شكرا

どうもありがとう *gracias*