Phase-field Solver with Dynamic Interface Refinement

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This is a two-phase two-component phase-field solver for OpenFOAM based on finite-volume discretization, with dynamically refining interface, and first order implicit Euler time integration scheme. The solver can be decomposed into multiple domains for parallel runs. To carry out the simulations, the steps mentioned below can be followed.

1 Isothermal undercooling

The first case is conducted by keeping the temperature constant and uniform throughout the melt.

1.1 Problem specification

The problem is defined as follows:

Solution domain The domain is considered to be two-dimensional in this problem. The square domain consists of a cylindrical undercooled region collocated with the centre of the square as shown in Figure 1.

Governing equations The basic idea of the phase-field model consists in the description of the evolution of individual phases using the corresponding set of order parameters (the phase-field variables). For each phase, the associated phase-field variable is equal to 1 inside the phase and vanishes to 0 outside the area occupied by the considered phase. The transition of phase fields from 1 to 0 at the phase boundaries is continuous and smooth, i.e., the phase boundaries are represented by diffuse interfaces with a finite thickness. Phase evolution is determined by the phenomenological minimization of the functional, which is formulated as the grand-potential functional (Choudhury and Nestler [(2012)]). The evolution equation for two phase-field variable can be written as:

$$\omega \epsilon \frac{\partial \phi}{\partial t} = \epsilon \left(\nabla \cdot \frac{\partial a(\nabla \phi)}{\partial \nabla \phi} \right) - \frac{1}{\epsilon} \frac{\partial w(\phi)}{\partial \phi} - \frac{\partial \Psi(T, \mu, \phi)}{\partial \phi}$$
 (1)

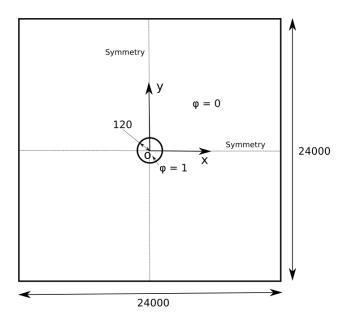


Figure 1: Geometry of the melt with an undercooled region.

• The double well potential $w(\phi)$ can be written as:

$$w(\phi) = \gamma \phi^2 (1 - \phi)^2 \tag{2}$$

• The gradient energy density, $a(\nabla \phi)$ has the form:

$$a(\nabla \phi) = \gamma [a_c(q)]^2 |q|^2 \tag{3}$$

where $q = |\nabla \phi|$ is the normal vector to the interface, γ controls the interfacial energy of the system and is known as surface energy density. $a_c(q)$ describes the anisotropy of evolving phase boundary. In case of four-fold symmetry $a_c(\nabla \phi) = 1 + \delta \cos(4\theta)$ where θ is the orientation of the interface and is computed in 2D as (Boettinger and Warren [(1996)]):

$$\tan \theta = \left(\frac{\partial \phi}{\partial y}\right) / \left(\frac{\partial \phi}{\partial x}\right) \tag{4}$$

where x and y define the coordinate system. Using above expression for θ , anisotropy can be represented as:

$$a_c(\nabla \phi) = 1 - \delta \left(3 - \frac{4[(\nabla_x \phi)^4 + (\nabla_y \phi)^4 + (\nabla_z \phi)^4]}{|\nabla \phi|^4} \right)$$
 (5)

where δ is the strength of anisotropy. $\nabla_x \phi$ is the x-component of $\nabla \phi$ and so on. The first term on the right hand side of evolution equation is then written as:

$$\frac{\partial a(\nabla \phi)}{\partial \nabla \phi} = 2\gamma a_c(\nabla \phi) \frac{\partial a_c(\nabla \phi)}{\partial \nabla \phi} |\nabla \phi|^2 + 2\gamma \nabla \phi [a_c(\nabla \phi)]^2$$
(6)

• The last term in equation (1) is related to the thermodynamic driving force. The grand potentials $\Psi(T,\mu)$ can be expressed as a linear expansion about the equilibrium chemical potential μ_{eq} :

$$\Psi_{\alpha}(T,\mu) = \Psi_{\alpha}(T,\mu_{eq}) + \frac{\partial \Psi_{\alpha}(T,\mu)}{\partial \mu}|_{\mu_{eq}}(\mu - \mu_{eq})$$
(7)

• The driving force ΔF^{α} is then written as:

$$\Delta F^{\alpha} = \left[\Psi_{\alpha}(T,\mu) - \Psi_{\beta}(T,\mu)\right] \frac{\partial h_{\alpha}(\phi)}{\partial \phi}
= \left(\frac{\partial \Psi_{\alpha}(T,\mu)}{\partial \mu}\Big|_{\mu_{eq}} - \frac{\partial \Psi_{\beta}(T,\mu)}{\partial \mu}\Big|_{\mu_{eq}}\right) (\mu - \mu_{eq}) \frac{\partial h_{\alpha}(\phi)}{\partial \phi}
= -\left[c^{\alpha}(T,\mu_{eq}) - c^{\beta}(T,\mu_{eq})\right] (\mu - \mu_{eq}) \frac{\partial h_{\alpha}(\phi)}{\partial \phi}$$
(8)

where $h(\phi)$ is an interpolation function given as:

$$h(\phi) = \phi^3 (6\phi^2 - 15\phi + 10) \tag{9}$$

• Growth of an undercooled melt starting from a small solid ($\phi = 1$) seed with appropriate levels of anisotropy typically exhibit dendritic expansion. Noise has been introduced at controlled levels to induce side branching in most simulations. This is typically implemented using random fluctuations of a source term added to the phase-field equation that is localized to regions where ϕ is between 0 and 1; i.e., in the interfacial region. This permits the inclusion of nucleation processes into simulations using the phase-field method (Boettinger and Warren [(1996)]). Thus, phase-field equation is modified by adding a noise term which is given as:

$$\frac{\partial \phi}{\partial t} \Longrightarrow \frac{\partial \phi}{\partial t} - 6rM\phi^2 (1 - \phi)^2 \tag{10}$$

where r is a random number distributed uniformly between 0 and 1, and a new number is generated for every point of the grid, at each time-step. M is a magnitude of noise which is taken as 0.003.

• The phase-field equation for two phases after incorporating above equation can be given as:

$$\omega \epsilon \frac{\partial \phi}{\partial t} = \epsilon \left(\nabla \cdot \frac{\partial a(\nabla \phi)}{\partial \nabla \phi} \right) - 18 \frac{\gamma}{\epsilon} \phi (1 - \phi) (1 - 2\phi)$$

$$+ \left[c^{\alpha}(T, \mu_{eq}) - c^{\beta}(T, \mu_{eq}) \right] (\mu - \mu_{eq}) \frac{\partial h_{\alpha}(\phi)}{\partial \phi} + 6rM\phi^{2} (1 - \phi)^{2}$$
(11a)

• In this model, direct solution is conducted for the thermodynamic variable μ , which relate the phase concentrations c_i^{α} . This is possible because the concentrations $c_i^{\alpha}(T,\mu)$ are written as explicit functions of the chemical potential μ . As will be seen in the solver source code, the equilibrium chemical potential, μ_{eq} is expressed in terms of equilibrium composition, c^{eq} , temperature difference with melting temperature, $(T_0 - T)$ and slope of the liquidus (or rate of change of temperature w.r.t. composition), m_{β} . According to the parabolic approximation for free energy:

$$\mu_{eq} = 2Ac^{eq} - 2A(T_0 - T)/m_{\beta} \tag{11b}$$

A is derived following the consequence of equilibrium between the different phases described by the common tangent construction.

• The evolution equation for μ can be written as:

$$\left(\frac{\partial c^{\alpha}(T,\mu)}{\partial \mu}h_{\alpha}(\phi) + \frac{\partial c^{\beta}(T,\mu)}{\partial \mu}[1 - h_{\alpha}(\phi)]\right)\frac{\partial \mu}{\partial t}
= \nabla \cdot \left[\left(D^{\alpha}g_{\alpha}(\phi)\frac{\partial c^{\alpha}(T,\mu)}{\partial \mu} + D^{\beta}[1 - g_{\alpha}(\phi)]\frac{\partial c^{\beta}(T,\mu)}{\partial \mu}\right)\nabla \mu\right] - \left[c^{\alpha}(T,\mu) - c^{\beta}(T,\mu)\right]\frac{\partial h_{\alpha}(\phi)}{\partial t}$$
(12)

where $c^{\alpha,\beta}(\mu)$ are the phase concentrations as functions of the independent chemical potential μ . D^{α} , D^{β} are the independent interdiffusivities in the two respective phases. It is noteworthy that this equation looks very similar to the evolution equation of the temperature field in pure materials. The last term on the right-hand

side $c^{\alpha}(\mu,T) - c^{\beta}(\mu,T)$ corresponds to a source term for rejection of mass at the interface during growth, which is analogous to the release of latent heat in pure material solidification. In this case diffusivity in solid phase is assumed to be negligible. Thus, $D^{\alpha} = 0$. This then becomes the case of one sided diffusion. For the case of one-sided diffusion, it has been shown in various previous works that there exists a thin-interface defect called solute trapping when simulations are performed with interface thicknesses, orders of magnitude larger than those of a real interface. The methodology proposed to correct this effect is the incorporation of an anti-trapping current in the evolution equation of the chemical potential. The anti-trapping term is incorporated as an additional flux of solute from the solid to the liquid in the normal direction to the interface (Karma [(2001)]). The modified evolution equation for the chemical potential along with the anti-trapping term is:

$$\left(\frac{\partial c^{\alpha}(T,\mu)}{\partial \mu}h_{\alpha}(\phi) + \frac{\partial c^{\beta}(T,\mu)}{\partial \mu}[1 - h_{\alpha}(\phi)]\right)\frac{\partial \mu}{\partial t}
= \nabla \cdot \left(D^{\beta}[1 - g_{\alpha}(\phi)]\frac{\partial c^{\beta}(T,\mu)}{\partial \mu}\nabla \mu - j_{at}\right) - \left[c^{\alpha}(T,\mu) - c^{\beta}(T,\mu)\right]\frac{\partial h_{\alpha}(\phi)}{\partial t}$$
(13a)

Further, $g_{\alpha}(\phi)$, the interpolation function is chosen as ϕ . The slope of c- μ curves is approximated according to the parabolic approximation for free energy in the solver source code:

$$\frac{\partial c^{\alpha}(T,\mu)}{\partial \mu} = \frac{\partial c^{\beta}(T,\mu)}{\partial \mu} = \frac{1}{2A}$$
 (13b)

• To make sure that the anti-trapping current appears in the first-order correction to the chemical potential, the anti-trapping current is formulated into the following form:

$$j_{at} = s(\phi)\epsilon[c^{\beta}(T,\mu) - c^{\alpha}(T,\mu)]\frac{\partial\phi}{\partial t}\frac{q_{\alpha\beta}}{|q_{\alpha\beta}|}$$
(14)

where $s(\phi)$ is a function such that the chemical potential jump vanishes at the interface. $q_{\alpha\beta}$ is the normal vector to the interface given by $(\phi_{\alpha}\nabla\phi_{\beta} - \phi_{\beta}\nabla\phi_{\alpha})$. For the case of only two phases, the expression of the anti-trapping current can be reduced to:

$$j_{at} = -s(\phi)\epsilon[c^{\beta}(T,\mu) - c^{\alpha}(T,\mu)]\frac{\partial\phi}{\partial t}\frac{\nabla\phi}{|\nabla\phi|}$$
(15)

• All terms in the above equation are used in the nondimensional form, so ϵ is the nondimensional parameter related to the interface width and t is the nondimensional time. $s(\phi)$ is a shape function which is taken as a constant value of $\frac{1}{2\sqrt{2}}$.

Boundary conditions

• Zero-flux is specified at all planes.

$Initial\ conditions$

- $\phi = 1.0$ and $\mu = 0.689$ inside the undercooled region.
- $\phi = 0.0$ and $\mu = 1.0$ outside the undercooled region.

Physical properties The nondimensionalized physical properties pertaining to AlZn alloy are:

- Slope liquidus $(m_{\beta}) = 0.45$
- Slope solidus $(m_{\alpha}) = 1e9$
- Relaxation coefficient (ω) = 1.687
- Surface energy $(\gamma) = 1$
- Interface width $(\epsilon) = 48$

- Composition of solid $(c^{\alpha}) = 0.78$
- Composition of liquid $(c^{\beta}) = 0.45$
- Equilibrium composition $(c^{eq}) = 0.5$
- Diffusivity in liquid $(D^{\beta}) = 1$
- Anti-trapping coefficient $(s(\phi)) = 0.35355$
- Thermal gradient (G) = 0
- Velocity (v) = 0
- Noise magnitude (M) = 0.003
- Constant value from temperature profile $(T_{initial}) = 0.93$
- Melting temperature $(T_0) = 1.0$
- Strength of anisotropy $(\delta) = 0.02$

The slope of the solidus (m_{α}) is kept very high for equation (11b) to be valid.

Solver name

• phaseFieldDynamic: An implementation with dynamic interface refinement for phase-field method to model solidification of two-phase two-component system.

Case name

• free Growth, located in the \$FOAM RUN/Phase Field Solver Dynamic directory.

The problem is solved using *phaseFieldDynamic*, located in the \$FOAM_RUN/PhaseFieldSolverDynamic directory. The code-guide for this solver is given in the \$FOAM_RUN/PhaseFieldSolverDynamic/codeGuide directory. Before proceeding further, it may be helpful for the user to become familiar with OpenFOAM documentation.

1.2 Pre-processing

To prepare case files and running the case, the user can change to the case directory: cd \$FOAM RUN/PhaseFieldSolverDynamic/freeGrowth

1.2.1 Mesh generation

OpenFOAM solves the case in three dimensions by default but can be instructed to solve in two dimensions. Here, the mesh must be 1 cell layer thick. Since this is quarter symmetric problem (Choudhury and Nestler [(2012)]), only the top-right quarter of the whole geometry (shown in Figure 1) is considered. The domain consists of a square of side length of 12000 in the x-y plane. Large variations in fields can be expected near the left and bottom boundaries, so the mesh will be graded to be smaller in this region. A graded mesh of 250 by 250 cells will be used initially. The block structure is shown in Figure 2.

The entries in blockMeshDict located in the system directory for this case are as follows:

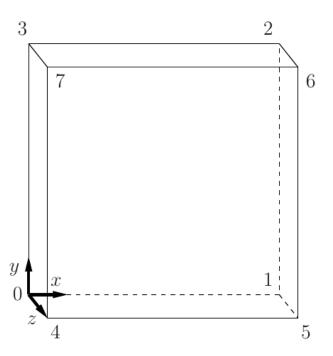


Figure 2: Block structure of the mesh.

```
2.0;
    version
    format
                ascii;
    class
                dictionary;
    object
                blockMeshDict;
}
convertToMeters 48;
x_min 0;
x_max 250; //direction of growth
y_min 0;
y_max 250;
z_min 0;
z_max 2;
// should all be integers
lx #calc "$x_max - $x_min";
ly #calc "$y_max - $y_min";
lz #calc "$z_max - $z_min"; // should all be integers
vertices
(
    ($x_min $y_min $z_min)
    ($x_max $y_min $z_min)
    ($x_max $y_max $z_min)
    ($x_min $y_max $z_min)
    ($x_min $y_min $z_max)
    ($x_max $y_min $z_max)
    (x_max y_max z_max)
    ($x_min $y_max $z_max)
);
```

```
blocks
(
   hex (0 1 2 3 4 5 6 7) ($lx $ly 1) simpleGrading (13 13 1)
// for dx = 0.5: simpleGrading (2 2 2) or convertToMeters = 0.5
edges
(
);
//====== FOR ZERO-FLUX BOUNDARY CONDITIONS ========
boundary
(
   floor
   {
        type symmetryPlane;
       faces
       (0\ 1\ 5\ 4)
       );
   }
   left
   {
       type symmetryPlane;
       faces
       (0 \ 4 \ 7 \ 3)
       );
   }
   {\tt sideSolid}
       type wall;
       faces
           (0 \ 3 \ 2 \ 1)
           (2 \ 3 \ 7 \ 6)
           (4567)
           (1 \ 2 \ 6 \ 5)
       );
   }
);
mergePatchPairs
(
);
The mesh is generated after accessing blockMesh entry within FOAM_RUN/PhaseFieldSolverDynamic/freeGrowth/Allrun:
#!/bin/sh
```

The logs are stored into log.blockMesh.

1.2.2 Boundary and initial conditions

Once the mesh generation is complete, the user can look at this initial conditions set up for this case. The case is set up to start at time t=0 s, so the initial field data is stored in a θ sub-directory of the *freeGrowth* directory. The θ sub-directory contains *phi.orig* and *mu.orig*, one for each of the phase parameter (ϕ) and chemical potential (μ) fields whose initial values and boundary conditions must be set. First, the file *phi.orig* can be examined:

```
/*----*\
\\/
      M anipulation |
FoamFile
{
        2.0;
  version
  format
        ascii;
  class
        volScalarField;
        "0";
  location
  object
        phi;
}
              dimensions
        [0 0 0 0 0 0 0];
internalField
        uniform 0;
//====== NO-FLUX BOUNDARY CONDITIONS ==========//
boundaryField
```

```
{
    floor
    {
                          symmetryPlane;
    }
    left
    {
        type
                          symmetryPlane;
    }
    sideSolid
    {
                          zeroGradient;
        type
    }
}
```

For this case, the boundary consists of symmetry Plane, split into 2 patches named: (1) floor for the bottom symmetric boundary; (2) left for the left symmetric boundary. The side Solid patch includes other planes of the 2D case and is given a zero Gradient boundary condition for ϕ , meaning "the normal gradient of ϕ is zero." The initial fields are set to uniform zero.

The user can similarly examine the chemical potential field in the $\theta/mu.orig$ file. The non-dimensionalized internal field is initialised as uniform zero. The boundary field for chemical potential requires the same boundary condition.

```
/*----*\ C++ -*-----*
          F ield | OpenFOAM: The Open Source CFD Toolbox Operation | Version: 4.x And | Web: www.OpenFOAM.org
\\
       / O peration
   \\ /
          M anipulation |
FoamFile
{
             2.0;
   version
   format
             ascii;
   class
             volScalarField;
             "0";
   location
   object
}
              [0 0 0 0 0 0]:
dimensions
internalField uniform 0;
//======= NO-FLUX BOUNDARY CONDITIONS =========//
boundaryField
   floor
   {
                     symmetryPlane;
   }
   left
```

```
{
    type symmetryPlane;
}
sideSolid
{
    type zeroGradient;
}
```

1.2.3 Setting initial field

A non-uniform initial condition is specified for the phase parameter, ϕ , where $\phi = 0$ for the liquid phase and 1 for the solid phase. And for chemical potential, μ , where $\mu = 1$ for the liquid phase and 0.689 for the solid phase.

This is achieved by running the setFields utility. It requires a *setFieldsDict* dictionary, located in the *system* directory, whose entries for this case are shown below:

```
/*----*\ C++ -*-----*\
| =======
\\/
        M anipulation |
\*-----*/
FoamFile
{
  version
          2.0;
  format
          ascii;
  class
          dictionary;
  location
          "system";
  object
          setFieldsDict;
}
defaultFieldValues
  volScalarFieldValue phi 0.0
  volScalarFieldValue mu 1.0
);
regions
(
  cylinderToCell
     p1 (0 0 00);
     p2 (0 0 200);
  radius 120;
     fieldValues
       volScalarFieldValue phi 1.0
```

The user can execute setFields through $FOAM_RUN/PhaseFieldSolverDynamic/freeGrowth/Allrun$. The logs are stored into log.setFields.

1.2.4 Physical properties

The physical properties for the case are specified in constant/transportProperties dictionary. The entries of transportProperties dictionary are shown below:

```
-----*\
| =======
| OpenFOAM: The Open Source CFD Toolbox
\\/
          M anipulation |
\*-----/
FoamFile
   version 2.0;
   format
           ascii;
           dictionary;
   class
   location
            "constant";
           transportProperties;
   object
}
dimx [0 1 0 0 0 0 0] 1; //Dimension of position
dimx
dimt
         dimt [0 0 1 0 0 0 0] 1; //Dimension of time
         m_0 [0 0 0 0 0 0 0] 1e9; //Slope solidus
m_0
         m_1 [0 0 0 0 0 0] 0.45; //Slope liquidus
m_1
         omega [0 0 0 0 0 0 0] 1.687; //Relaxation coefficient for phi (order parameter)
omega
         gamma [0 0 0 0 0 0 0] 1; //Surface Energy
gamma
         epsilon [0 0 0 0 0 0 0] 48; //Interface Width
epsilon
c_Sol
         c_Sol [0 0 0 0 0 0 0] 0.78; //Composition of solid in equilibrium with liquid
         c_Liq [0 0 0 0 0 0 0] 0.45; //Composition of liquid in equilibrium with solid
c_Liq
         c_eq [0 0 0 0 0 0 0] 0.5; //Equilibrium composition or average composition of alloy
c_eq
         diff Sol [0 0 0 0 0 0 0] 1; //Diffusivity in solid
diff Sol
diff_Liq
         diff_Liq [0 0 0 0 0 0 0] 1; //Diffusivity in liquid
anti trap
         anti trap [0 0 0 0 0 0] 0.35355; //Antitrapping coefficient
G
         G [0 0 0 0 0 0 0] 0.0; //Thermal gradient
         v [0 0 0 0 0 0 0] 0.0; //Velocity
                noise_mag [0 0 0 0 0 0 0] 0.003; //Noise magnitude
noise_mag
initial
              initial [0 0 0 0 0 0 0] 0.93; //Constant value from temperature profile
          TO [0 0 0 0 0 0 0] 1.0; //Melting temperature
T0
          TO [0 0 0 0 0 0] 0.93; //Operating temperature
             delta_01 [0 0 0 0 0 0 0] 0.02; //Strength of anisotropy
delta_01
```

For this case, thermal gradient, G and pulling velocity, v are kept zero to allow solidification in the form of freely growing dendrites. The physical properties are read by readTransportProperties.H while running the case.

1.2.5 Time step control

Input data relating to the control of time, reading and writing of the solution data are read from the *controlDict* dictionary located in the *system* directory.

The run is started at time t = 0 to read field data from a directory named θ . Therefore, the startFrom keyword is set to startTime and then the startTime keyword to be 0.

The end time can be considered to be the time taken for the primary dendrites to reach the boundaries, which is found to be 3000000 for this case. To specify this end time, the stopAt keyword is set to endTime and then the endTime keyword to 3000000.

Next, the time step must be fixed which is represented by the keyword deltaT. To achieve temporal accuracy and numerical stability while reducing computational effort, deltaT is set to 200.

As the simulation progresses, results written at certain intervals of time can later be viewed with a post-processing package. The writeControl keyword can be set to the runTime option, where the time interval for writing is specified under the writeInterval keyword. The writeInterval can be set to 100000. OpenFOAM creates a new directory named after the current time, e.g. 500000, on each occasion that it writes a set of data containing the results for each field, phi and mu, into the time directories. For this case, the entries in the controlDict are shown below:

```
/*----*\
                        | OpenFOAM: The Open Source CFD Toolbox
          F ield
                        | Version: 4.0
           0 peration
   \\ /
           A nd
                        | Web:
                                   www.OpenFOAM.org
    \\/
           M anipulation |
FoamFile
{
              2.0;
   version
   format
              ascii;
   class
              dictionary;
   location
              "system";
   object
              controlDict;
application
              phaseFieldDynamic;
startFrom
              startTime;
startTime
              0; // the actual time i.e. runtime.value() or no. of iterations*deltaT
stopAt
              endTime;
endTime
              3000000; //actual time = total_iterations*deltaT
```

```
deltaT
                200;
writeControl
                runTime;
writeInterval
                100000;
purgeWrite
                0;
writeFormat
                ascii;
writePrecision 6;
writeCompression off;
timeFormat
                general;
timePrecision
runTimeModifiable true;
```

1.2.6 Discretisation schemes

In this case, the term $\left(\nabla \cdot \frac{\partial a(\nabla \phi)}{\partial \nabla \phi}\right)$ in the phase-field equation can be decomposed into laplacian(ac_01*ac_01,phi) and div(dadgradPhi) (see alphaEqn.H). The term $\nabla \cdot \left(\left[1-g_{\alpha}(\phi)\right]\frac{\partial c^{\beta}(T,\mu)}{\partial \mu}\nabla \mu - j_{at}\right)$ in the chemical potential equation includes laplacian((1-phi),mu). The laplacian and divergence are solved by fixing laplacianSchemes and divSchemes to Gauss linear corrected and Gauss linear, respectively.

The other discretised terms use commonly employed schemes so that the fvSchemes dictionary entries should therefore be:

```
| OpenFOAM: The Open Source CFD Toolbox
| \\
        / F ield
                       | Version: 4.0
| Web:
   \\ /
           A nd
                                  www.OpenFOAM.org
    \\/
           M anipulation |
FoamFile
   version
             2.0;
   format
             ascii;
   class
              dictionary;
              "system";
   location
             fvSchemes;
   object
}
ddtSchemes
{
   default
                 Euler;
```

```
gradSchemes
   default
              Gauss linear;
   grad(phi)
                Gauss linear;
}
divSchemes
{
   default
               Gauss linear;
}
laplacianSchemes
   default
               Gauss linear corrected;
}
\verb|interpolationSchemes||
   default
               linear;
}
{\tt snGradSchemes}
   default
              corrected;
}
// **********************************//
1.2.7 Solver settings
In the fvSolution, the solver tolerance should be set to 10^{-6} for this case. The solver relative tolerance,
denoted by relTol, is set to 0.
/*----*\
| OpenFOAM: The Open Source CFD Toolbox
| Version: 4.0
 | Web:
                               www.OpenFOAM.org
  \\/ M anipulation |
```

```
14
```

 ${\tt FoamFile}$

solvers
{

version

location

format

class

object

2.0;

ascii;

dictionary;

fvSolution;

"system";

```
mu
   {
       solver
                              smoothSolver;
                               DIC;
       preconditioner
       smoother
                             symGaussSeidel;
                              1e-6;
       tolerance
       relTol
                               0.0:
   }
   phi
   {
                               smoothSolver;
       solver
       preconditioner
                               DIC;
       smoother
                             symGaussSeidel;
                              1e-6;
       tolerance
       relTol
                               0.0;
   }
}
SIMPLE
{
   nNonOrthogonalCorrectors 0;
}
// ********************************//
```

The fvSolution dictionary contains a sub-dictionary, SIMPLE that contains a control parameter nNonOrthogonalCorrectors set to 0. The description of other options can be found in the OpenFOAM userguide.

1.3 Running the application

The first step to run a parallel case is to decompose the domain using the decomposePar utility for assigning to different processors. The dictionary associated with decomposePar, decomposeParDict is located in the system directory. The first entry is numberOfSubdomains corresponding to the number of processors available for the case. The method of decomposition can be simple. The domain is split into subdomains, in the x, y and z directions, the number of subdomains in each direction being given by the vector n. As this geometry is two dimensional, the 3rd direction, z, cannot be split, hence n_z must be 1. The n_x and n_y components of n split the domain in the x and y directions following $n_x n_y = \text{numberOfSubdomains}$. To minimize the communication time between the processors, the number of cell faces adjoining the subdomains are kept to a minimum. For a square geometry, the split between the x and y directions should be fairly even. The delta keyword is set to 0.001.

For this case, number Of Subdomains = 4 and n = (2, 2, 1). When executing *Allrun*, decompose Par is run. The logs are stored into log.decompose Par.

```
format
                ascii;
    class
                dictionary;
    note
                "mesh decomposition control dictionary";
                "system";
    location
    object
                decomposeParDict;
}
numberOfSubdomains
//- Keep owner and neighbour on same processor for faces in zones:
// preserveFaceZones (heater solid1 solid3);
//method
                  scotch;
method
                simple;
simpleCoeffs
    n
                 (2\ 2\ 1);
    delta
                0.001;
}
```

After compiling the solver (see *codeGuide*), it is executed parallely using runParallel within *Allrun*. The progress of the job is stored into *log.phaseFieldDynamic*. It includes the current time, initial and final residuals for all fields.

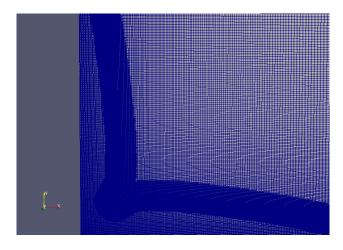


Figure 3: Refined interface at time 1000000.

The dynamicInterfaceRefineFvMesh (Holzmann [(2017)]) can be used to set refinement level for two independent ranges during run time. This allows additional flexibility of setting very fine mesh at the interface. The mesh near the interface is kept finer than the mesh far away from the interface as shown in Figure 3. This is done by introducing buffer layers, which adds a set number of layer usually between 2-5 while transitioning from one refinement level to the other. Additionally, in meshes with more than two refinement levels, it is important to ensure a smooth transition between the different levels. This is done to sufficiently decrease discretization errors due to mesh distortion at refinement transitions, and to provide a buffer between two refinement levels for the computed flow to adapt to the new mesh level. The library libdynamicInterfaceRefineFvMesh.so (see codeGuide) is called from constant/dynamicMeshDict which is shown below:

```
-----*\
                    \\/ M anipulation |
\*-----/
FoamFile
{
  version 2.0;
          ascii;
  format
  class dictionary;
location "constant";
  dynamicFvMeshLibs ( "libdynamicInterfaceRefineFvMesh.so" );
dynamicFvMesh dynamicInterfaceRefineFvMesh;
{\tt dynamicInterfaceRefineFvMeshCoeffs}
{
  // Field to be refined
  field
        phi;
  // Setting for Interface handling
      // How often to refine
      refineInterface 50;
      // Refine field inbetween lower..upper
      lowerRefineLevelInterface 0.001;
      upperRefineLevelInterface 0.999;
      // If unrefineLevelMin > value of field -> unrefine
      // If unrefineLevelMax < value of field -> unrefine
      unrefineLevelInterfaceMin 0.0001;
      unrefineLevelInterfaceMax
      // Have slower than 2:1 refinement
     nBufferLayersInterface 1;
      // Refine cells only up to maxRefinement levels
      maxRefinementInterface 2;
      // Stop refinement if maxCells reached
      maxCellsInterface
                    2000000;
   // Setting for Phase handling
      // How often to refine
      refinePhase 1;
      // Refine field inbetween lower..upper
```

```
lowerRefineLevelPhase 0.000000001;
       upperRefineLevelPhase 0.001;
       // If unrefineLevelPhase > value of field -> unrefine
       unrefineLevelPhase
                           -1.10;
       // Have slower than 2:1 refinement
       nBufferLayersPhase
                           0:
       // Refine cells only up to maxRefinement levels
       maxRefinementPhase
                           1;
       // Stop refinement if maxCells reached
       maxCellsPhase
                           20000000;
   // Flux field and corresponding velocity field. Fluxes on changed
   // faces get recalculated by interpolating the velocity. Use 'none'
   // on surfaceScalarFields that do not need to be reinterpolated.
   correctFluxes
   );
   // Write the refinement level as a volScalarField
   dumpLevel
                   false;
}
// ********************************//
```

When dynamic interface refinement is used in parallel, the processor handling the subdomain with refining interface will have the largest load. The overall computation becomes inefficient when processed data is required by the waiting processors. The size of directories generated from each processors can be used to estimate the load imbalance. The larger the variation in directory sizes from processors, greater is the load imbalance. Due to unavailability of any dynamic load balancing library in OpenFOAM, a domain decomposition method, scotch can be used. Scotch decomposition does not require any geometrical parameter input, and attempts to minimize the number of processor boundaries. Hence, the standard deviation of number of cells per processor can be minimized. It must be mentioned that load balancing involves stopping and restarting the simulation (creating the mesh, and allocating all the cells to processors), and thereby involves a fairly high CPU time. Therefore, to overcome the challenge of load balancing between the processors due to dynamic refinement, a dynamic load balancing library can be important in OpenFOAM.

1.4 Post-processing

Once the case has completed running, the decomposed fields and mesh from processor directories can be reconstructed for post-processing. Firstly, the common point data between the neighboring processors must be generated using the reconstructParMesh utility. The tolerance of reconstructParMesh can be specified using mergeTol, which must match with writePrecision in *controlDict*. Then using reconstructPar, the decomposed mesh with fields can be reassembled. This can be done by executing *construct* which is shown below:

```
reconstructParMesh -mergeTol 1e-06
reconstructPar
rm -r ./processor*/
```

The results reconstructed to time directories can be viewed using ParaView by creating and opening free Growth.foam case module:

cd \$FOAM_RUN/PhaseFieldSolverDynamic/freeGrowth

 $touch\ free Growth. foam$

paraview freeGrowth.foam

The evolution of phase-field profile corresponding to three time steps are shown in Figure 4-6.

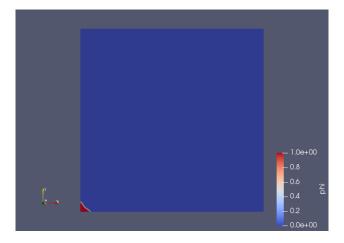


Figure 4: Phase-field profile at time 100000.

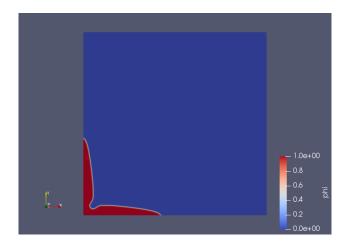


Figure 5: Phase-field profile at time 1000000.

As can be seen from the figures, in absence of thermal gradient, the solidification happens in the form of freely growing dendrites (Choudhury and Nestler [(2012)]). Also, the chemical potential plot at time 2500000 is shown in Figure 7.

2 Directional solidification

The second case is performed by subjecting the melt to a positive thermal gradient ahead of the solid-liquid interface. The imposed temperature field as a function of thermal gradient in the x direction, G and pulling velocity, v is taken as:

$$T(x,t) = T_{initial} + G(x - vt)$$
(16)

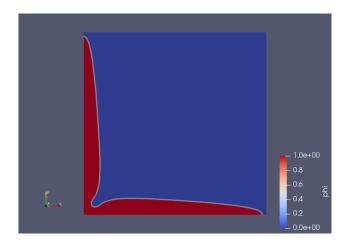


Figure 6: Phase-field profile at time 2500000.

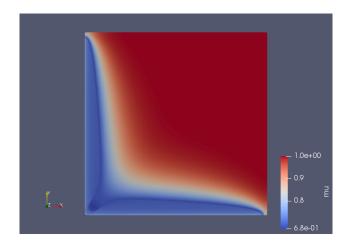


Figure 7: Chemical potential at time 2500000.

2.1 Problem specification

Solution domain The domain is considered to be two-dimensional in this problem. The rectangular domain consists of an undercooled region as shown in Figure 8.

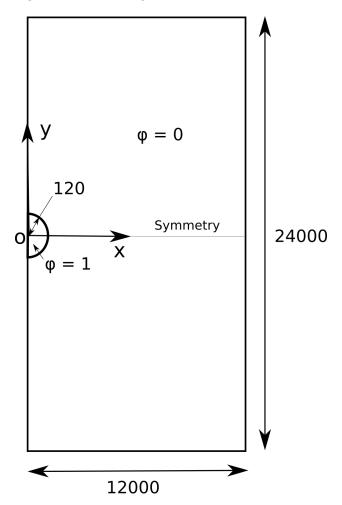


Figure 8: Geometry of the melt with an undercooled region.

Governing equations The governing equations used are same as in section 1.1.

Boundary conditions Zero-flux is specified at all planes.

Initial conditions The initial conditions are same as in section 1.1.

Physical properties The physical properties are same as in section 1.1 except the following (in order to impose thermal gradient and velocity in the x direction):

- Thermal gradient (G) = 1.7e-5
- Velocity (v) = 0.001

Solver name phase Field Dynamic: An implementation with dynamic interface refinement for phase-field method to model solidification of two-phase two-component system.

 ${\it Case \ name \ directional Solidification}, \ {\it located \ in \ the \ \$FOAM_RUN/Phase Field Solver Dynamic \ directory}.$

2.2 Pre-processing

To prepare case files and running the case, the user can change to the case directory: cd \$FOAM_RUN/PhaseFieldSolverDynamic/directionalSolidification

2.2.1 Mesh generation

The overall process of generating the mesh remains same as in section 1.2.1. However, this case is a symmetric problem. So, only the top half of the whole geometry (shown in Figure 8) is considered. Similarly, graded mesh of 250 by 250 cells (with smaller cells near the left and bottom boundaries) will be used. The entries in blockMeshDict located in the system directory for this case are as follows:

```
/*----*\
\*-----/
FoamFile
{
  version 2.0; format ascii;
         dictionary;
  class
  object
         blockMeshDict;
}
convertToMeters 48;
x_{min} 0;
x_max 250; //direction of growth
y_min 0;
y_max 250;
z_min 0;
z max 2;
// should all be integers
lx #calc "$x_max - $x_min";
ly #calc "$y_max - $y_min";
lz #calc "$z_max - $z_min"; // should all be integers
vertices
(
  ($x_min $y_min $z_min)
  ($x_max $y_min $z_min)
  ($x_max $y_max $z_min)
  ($x_min $y_max $z_min)
  ($x_min $y_min $z_max)
  ($x_max $y_min $z_max)
  ($x_max $y_max $z_max)
  ($x_min $y_max $z_max)
);
blocks
  hex (0 1 2 3 4 5 6 7) ($lx $ly 1) simpleGrading (13 13 1)
// for dx = 0.5: simpleGrading (2 2 2) or convertToMeters = 0.5
```

```
);
edges
(
);
//====== FOR ZERO-FLUX BOUNDARY CONDITIONS =======
boundary
(
   floor
   {
        type symmetryPlane;
       faces
       (0\ 1\ 5\ 4)
       );
   }
   left
       type wall;
       faces
       (0 4 7 3)
       );
   }
   {\tt sideSolid}
       type wall;
       faces
           (0 \ 3 \ 2 \ 1)
           (2 \ 3 \ 7 \ 6)
           (4 5 6 7)
           (1 \ 2 \ 6 \ 5)
       );
   }
);
mergePatchPairs
);
// *********************************//
```

2.2.2 Boundary and initial conditions

Similar to section 1.2.2, the file phi.orig in θ sub-directory of the directional Solidification directory can be examined:

```
O peration
                       | Version: 4.x
   \\ /
          A nd
                       | Web:
                                 www.OpenFOAM.org
          M anipulation |
FoamFile
{
   version
             2.0;
   format
             ascii;
   class
             volScalarField;
   location
             "0";
   object
             phi;
}
              dimensions
             [0 0 0 0 0 0 0];
internalField uniform 0;
//====== NO-FLUX BOUNDARY CONDITIONS =========//
boundaryField
   floor
   {
                    symmetryPlane;
      type
   }
   left
   {
                    zeroGradient;
      type
   }
   sideSolid
   {
                    zeroGradient;
      type
   }
}
```

For this case, the boundary consists of symmetry Plane, only for patch named floor for the bottom symmetric boundary. The remaining planes of the 2D case are given a zero Gradient boundary condition.

The boundary field for chemical potential requires the same boundary condition.

```
/*----*\
| =======
| OpenFOAM: The Open Source CFD Toolbox
| Version: 4.x
  \\ /
       A nd
                 | Web:
                        www.OpenFOAM.org
  \\/
       M anipulation |
FoamFile
{
  version
         2.0;
  format
         ascii;
  class
         volScalarField;
         "0";
  location
```

```
object
}
               [0 0 0 0 0 0 0];
internalField uniform 0:
//====== NO-FLUX BOUNDARY CONDITIONS ==========//
boundaryField
   floor
                      symmetryPlane;
       type
   }
   left
   {
                    zeroGradient;
       type
   sideSolid
                      zeroGradient;
       type
   }
}
```

2.2.3 Setting initial field

The entries of setFieldsDict dictionary for this case are kept same as section 1.2.3.

2.2.4 Physical properties

The entries of *transportProperties* dictionary are shown below:

```
/*----*\
| =======
\\/
     M anipulation \mid
FoamFile
 version 2.0;
 format
      ascii;
      dictionary;
 class
      "constant";
 location
 object
       transportProperties;
}
dimx [0 1 0 0 0 0 0] 1; //Dimension of position
dimx
     dimt [0 0 1 0 0 0 0] 1; //Dimension of time
dimt
     m_0
```

```
m 1 [0 0 0 0 0 0] 0.45; //Slope liquidus
m_1
           omega [0 0 0 0 0 0 0] 1.687; //Relaxation coefficient for phi (order parameter)
omega
           gamma [0 0 0 0 0 0 0] 1; //Surface Energy
gamma
           epsilon [0 0 0 0 0 0 0] 48; //Interface Width
epsilon
           c_{Sol} [0 0 0 0 0 0 0] 0.78; //Composition of solid in equilibrium with liquid
c_Sol
           c Liq [0 0 0 0 0 0 0] 0.45; //Composition of liquid in equilibrium with solid
c Liq
           c eq [0 0 0 0 0 0] 0.5; //Equilibrium composition or average composition of alloy
c_eq
diff_Sol
           diff_Sol [0 0 0 0 0 0 0] 1; //Diffusivity in solid
diff_Liq
           diff_Liq [0 0 0 0 0 0 0] 1; //Diffusivity in liquid
           anti_trap [0 0 0 0 0 0 0] 0.35355; //Antitrapping coefficient
anti_trap
           G [0 0 0 0 0 0 0] 1.7e-5; //Thermal gradient
           v [0 0 0 0 0 0 0] 0.001; //Velocity
                  noise_mag [0 0 0 0 0 0 0] 0.003; //Noise magnitude
noise_mag
                initial [0 0 0 0 0 0] 0.93; //Constant value from temperature profile
initial
T0
            T0 [0 0 0 0 0 0 0] 1.0; //Melting temperature
Т
            TO [0 0 0 0 0 0 0] 0.93; //Operating temperature
               delta_01 [0 0 0 0 0 0] 0.02; //Strength of anisotropy
delta_01
            A [0 0 0 0 0 0 0] 1;
//B
             B [0 0 0 0 0 0 0] 1;
```

It must be noted that, thermal gradient, G and velocity, v are nonzero for directional solidification. The remaining entries are same as section 1.2.4.

2.2.5 Time step control

Similar to section 1.2.5, the startTime and deltaT are specified. The endTime keyword is set to 6000000 to observe the growth of secondary dendrites from the primary dendrite, growing normal to the direction of imposed thermal gradient. The writeInterval can be set to 200000. For this case, the entries in the *controlDict* are shown below:

```
/*----*\
\\ /
        A nd
                 | Web:
                          www.OpenFOAM.org
        M anipulation |
   \\/
FoamFile
{
  version
          2.0:
  format
          ascii;
  class
         dictionary;
         "system";
  location
  object
          controlDict;
}
application
          phaseFieldDynamic;
startFrom
          startTime;
startTime
          0; // the actual time i.e. runtime.value() or no. of iterations*deltaT
stopAt
          endTime;
```

```
endTime
              6000000; //actual time = total_iterations*deltaT
deltaT
              200;
writeControl
              runTime;
              200000;
writeInterval
purgeWrite
              0;
writeFormat
              ascii;
writePrecision 6;
writeCompression off;
timeFormat
              general;
timePrecision
runTimeModifiable true;
// ********************************//
```

2.2.6 Discretisation schemes and solver settings

The fvSchemes and fvSolution dictionary entries are kept same as section 1.2.6 and 1.2.7.

2.3 Running the case

For this case, the primary dendrite can be expected to grow in the y direction since the thermal gradient is applied in the x direction. Consequently, the secondary dendrites will grow almost uniformly in the x direction (as will be seen later in section 2.5). In order to balance the load between the processors, in the presence of dynamic interface refinement and without stopping the run unlike section 1.3, the subdomains are divided in the y direction. Hence, number Of Subdomains = 4 and n = (1, 4, 1) are set.

```
-----*\
| =======
                    | OpenFOAM: The Open Source CFD Toolbox
| \\
     / F ield
                    | Version: 4.0
      / O peration
  \\ /
                    | Web:
                            www.OpenFOAM.org
         A nd
   \\/
         M anipulation |
         ·
-----*/
FoamFile
{
           2.0;
  version
  format
           ascii;
  class
           dictionary;
           "mesh decomposition control dictionary";
  note
           "system";
  location
  object
           decomposeParDict;
}
```

Similar to section 1.3, the solver for this case is run parallely using runParallel within Allrun.

2.4 Post-processing

Similar to section 1.4, the results reconstructed to time directories can be viewed by creating and opening the directional Solidification. foam case module using ParaView. The evolution of phase-field profile corresponding to three time steps are shown in Figure 9-11.

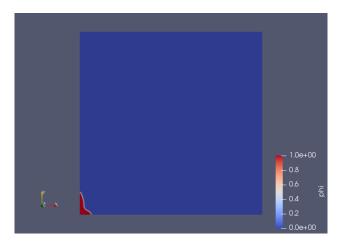


Figure 9: Phase-field profile at time 200000.

As can be seen from the figures, in the presence of thermal gradient and velocity in the x direction, the primary dendrite grows in the y direction. Later, secondary dendrites start to grow from the primary dendrite unlike the *freeGrowth* case in section 1.4. Also, the chemical potential plot at time 3600000 is shown in Figure 12.

References

[Boettinger and Warren(1996)] W J Boettinger and J A Warren. The phase-field method: Simulation of alloy dendritic solidification during recalescence. *Metall Mater Trans A*. 27:657–669, 1996.

[Choudhury and Nestler(2012)] A Choudhury and B Nestler. Grand-potential formulation for multicomponent phase transformations combined with thin-interface asymptotics of the double-obstacle potential. *Physical review. E, Statistical, nonlinear, and soft matter physics.* 85:021602, 2012.

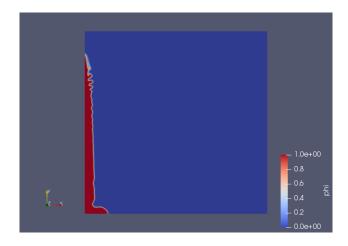


Figure 10: Phase-field profile at time 800000.

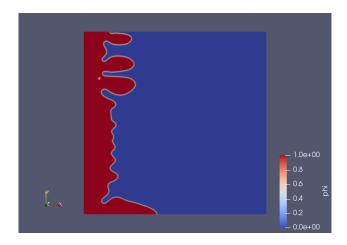


Figure 11: Phase-field profile at time 3600000.

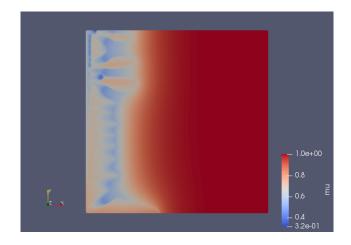


Figure 12: Chemical potential at time 3600000.

 $[Holzmann (2017)] \ T \ Holzmann. \ Https://bitbucket.org/shor-ty/dynamic$ $interfacerefinefvmesh/src/master/. \\ [Karma (2001)] \ A \ Karma. \ Phase-field formulation for quantitative modeling of alloy solidification. \ Phys. \ Rev. \\ Lett. \ 87:115701, \ 2001.$