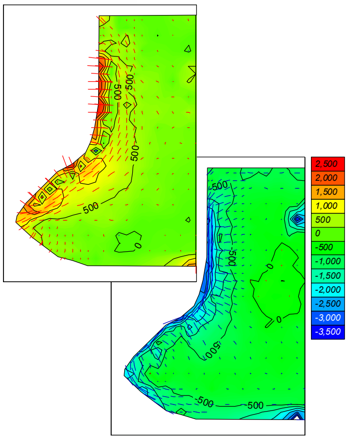
Engineering

Advanced diffraction techniques for Residual Stresses determination



Dr Michael Preuss
University of Manchester

Introduction

- Residual stresses in materials
- Principles of measuring residual stresses by diffraction
- Neutron and Synchrotron X-ray diffraction
  - Properties
  - Facilities
- Case Studies / Questions
- Conclusions

What are residual stresses

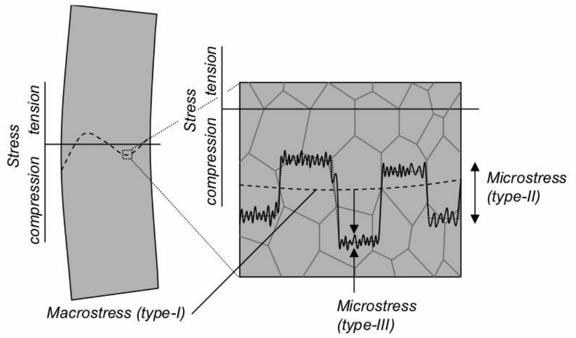
Deformation mismatch

Example: Welding

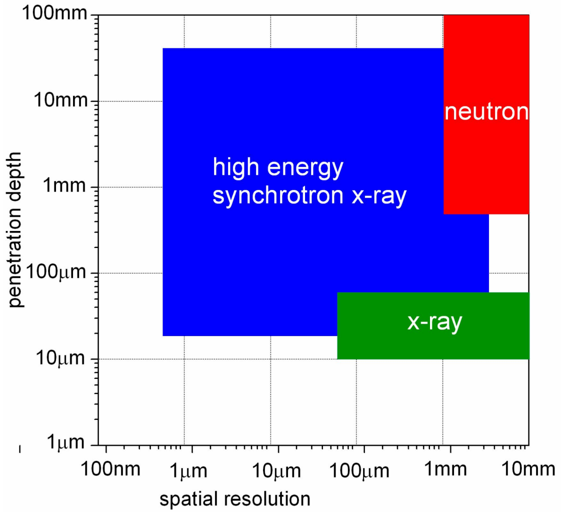
Residual Stresses

- Internal stresses
- Caused by misfit
  - Type I
  - Type II
  - Type III

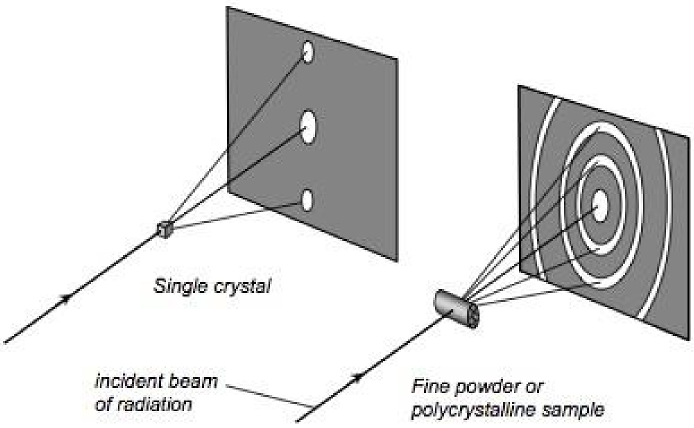
Bent bar:



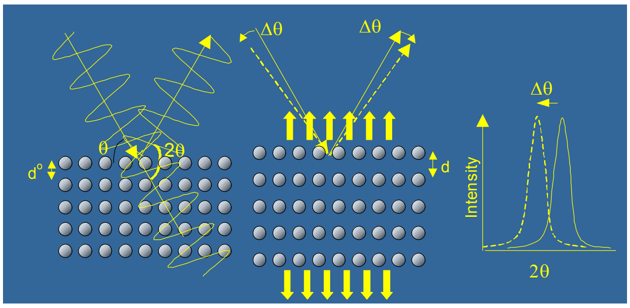
General Overview: Diffraction methods available



General Overview: Basic Principles: diffraction



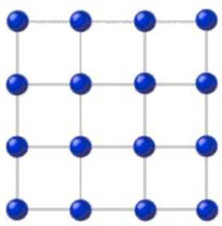
General Overview:
Basic Principles: diffraction



$$\lambda = 2d\sin\theta$$

- Diffraction measures elastic lattice strain as peak shifts
- Uses the poly-crystalline lattice planes as internal strain gauges

Basic Principle



Bragg scattering angle

X-ray wavelength: $\lambda = \frac{hc}{E} = \frac{12.39}{[keV]}$

Bragg's law: $\lambda = 2d \sin\theta$

$$\frac{12.39}{[keV]} = 2d \sin\theta$$

Scattering Angle

$$\theta = \arcsin\left(\frac{12.39}{2d[keV]}\right)$$

General Overview: Basic Principles

- Measured strains have to be converted into stresses! (Hooke's law) $\varepsilon = \frac{a - a_0}{a_0} = \frac{d - d_0}{d_0}$
- Often requires the unstrained lattice parameter a_0

e.g. isotropic triaxial along principal directions:

$$\varepsilon_{11} = \frac{1}{E}\left[\sigma_{11} - \upsilon(\sigma_{22} + \sigma_{33})\right]$$

$$\varepsilon_{22} = \frac{1}{E}\left[\sigma_{22} - \upsilon(\sigma_{33} + \sigma_{11})\right]$$

$$\varepsilon_{33} = \frac{1}{E}\left[\sigma_{33} - \upsilon(\sigma_{11} + \sigma_{22})\right]$$

(Attention: not always this simple!)

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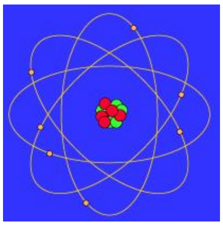
Why do we like neutrons ?

- Part of the nucleus
- Same mass as protons
- Interesting wavelength/mass relationship:

$$\lambda = \frac{h}{p} = \frac{h}{mv}$$

Planck

Mass * Velocity



- "Thermal" neutrons: wavelength similar to those of X-rays 0.5-5Å similar to atomic spacing in solids
- Allows cubic gauge volumes!
- Relatively divergent beam !!

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Neutron and Synchrotron Sources

Neutron:

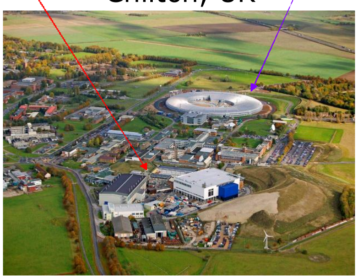
- Reactor Sources (Fission)
  - Constant wavelength/Single Peak
- Accelerator Sources (Spallation)
  - Time-of-flight / Full Spectra / Rietveld

Synchrotron:

- Monochromatic λ and white beam

ESRF Grenoble, France ILL

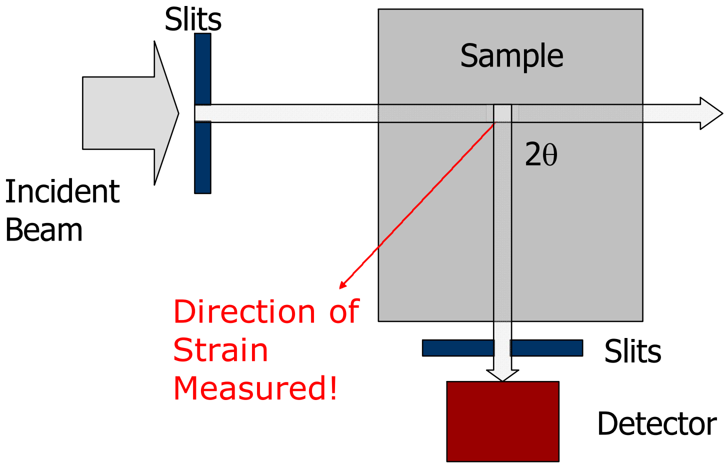
ISIS Chilton, UK Diamond



General Overview: Strain Scanning

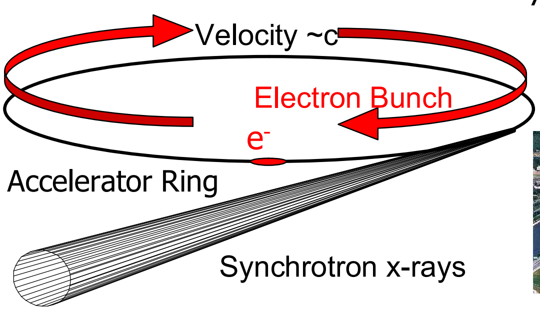
Diffracting Gauge Volume: The volume element defined by the incident slits and diffraction slits

Neutron Diffraction

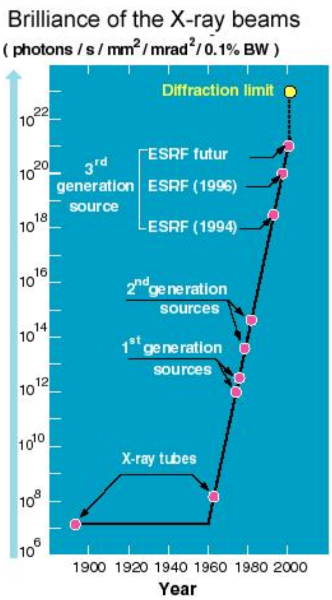


In comparison: Synchrotrons

- Synchrotron X-rays are energy X-rays (10-300keV) produced using "synchrotron" accelerators
- Main difference is the wavelength/energy
- Penetration depends on wavelength
- Very high intensity:
  - X-ray tube ~10E8
  - Synchrotron ~10E15



In comparison: Synchrotrons



- Synchrotron sources provide very intense (million times more flux than a lab source) high energy beams
- beam is highly parallel (10^{12} times more brilliance than a lab source)
- at energies of 40-80keV penetrations of many mm possible
- small micron sized beams

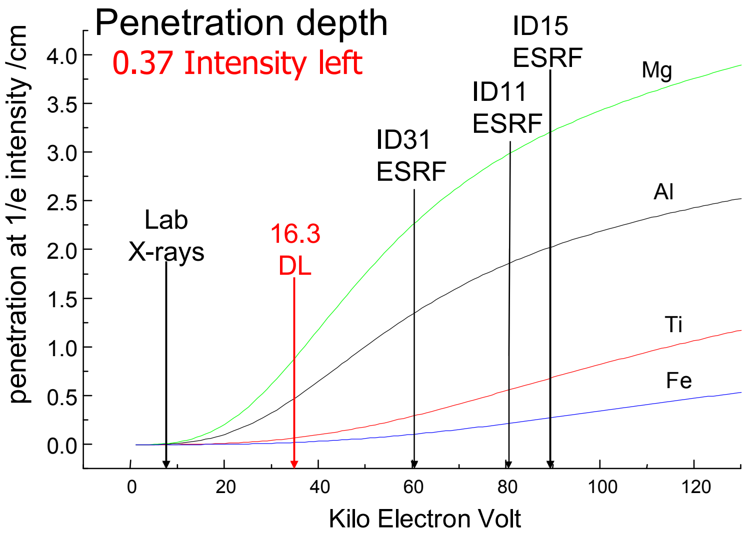
How does a Synchrotron work ?

- X-rays produced by sending high energy (9GeV) electrons round a ring
  - radial acceleration causes emission of electromagnetic radiation
  - low energy radiation
  - much greater radiation if you insert devices which bend beam sharply
- X-ray beam produced by bending magnets, undulators and wigglers

How does a Synchrotron work ?

- Bending magnets create a wide spectrum of X-ray radiation
- Wigglers are more intense because bend beam many times
- Undulators bend the beam such that radiation interferes to create very high fluxes of certain wavelengths (determined by spacing and number of magnets)
- Highly parallel beam is produced

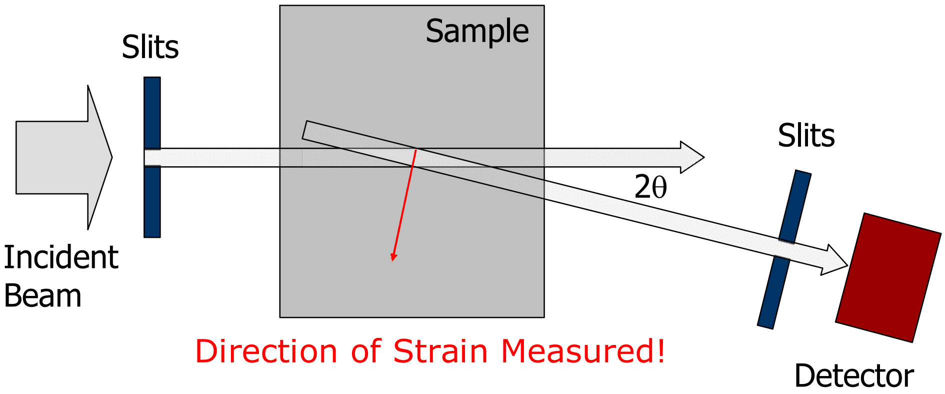
Synchrotron Diffraction: Penetration depth (monochromatic beam)



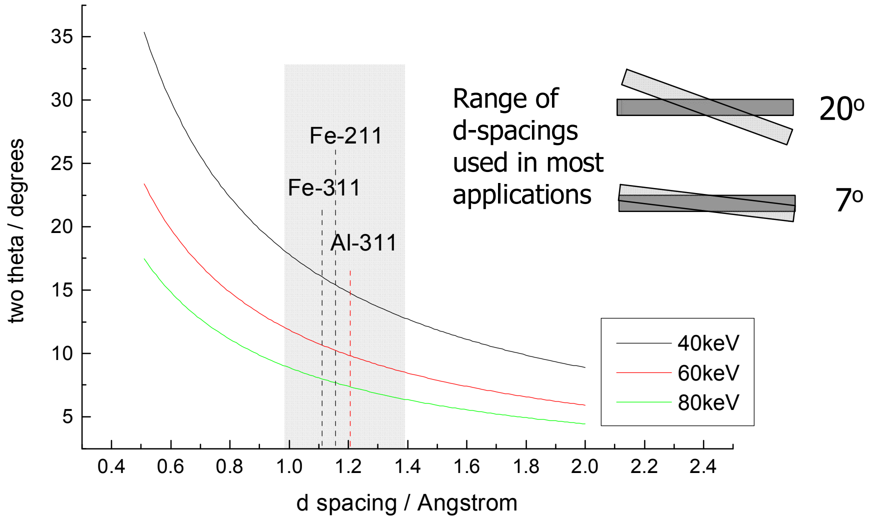
General Overview: Strain Scanning

Diffracting Gauge Volume: The volume element defined by the incident slits and diffraction slits

High Energy Synchrotron X-ray Diffraction



Synchrotrons: Scattering Angle 2θ



General Overview: Diffracting Gauge Volume

Volume element of the material in which the recorded scattering takes place

- Results in averaged d-spacing (powder diffraction - many grains)
- Defines the minimum spatial resolution of the method (around 1mm^3 minimum gauge volume when using neutron diffraction)
- and type of residual stress resolved (macro-stress or type-I usually. Type-II for two phase materials).
- Use the largest possible gauge volume for your specific issue in order to minimise counting time

Near surface measurements

Neither peak shift (strain) nor measurement location is correct near a surface!

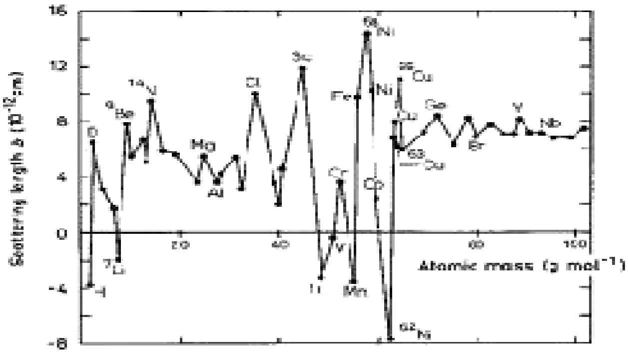
- Partial filling of sampling gauge gives a peak shift - need to correct peak shift
- Translator records centre of gauge which is rarely the centre of gravity of diffracting region
  - need to correct gauge position

Neutron Properties

- Neutrons are scattered by atomic nuclei (electrons and X-rays which are scattered by the electron cloud).
- Since the scattering is nuclear process, scattering amplitude varies greatly for different isotopes of same element and in a unpredictable manner from element to element. X-ray and electron scattering increase monotonically with atomic number

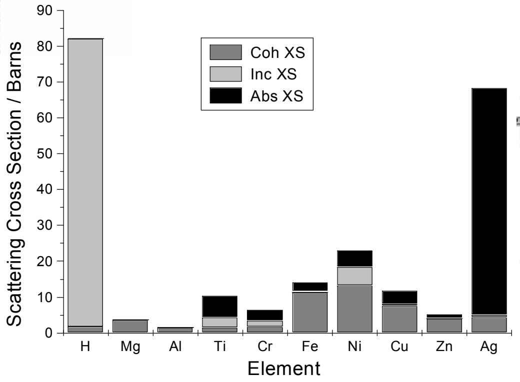
Neutron Properties

- Random Scattering length
- Penetration depth independent of energy/wavelength
- Electrically neutral
- Great penetration
- Low flux/intensity

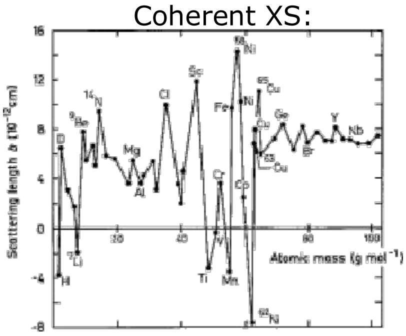


Economic Depth	Al	Steel	Cu	Ti	Ni	SiC
mm	250	37	40	27	24	200

Neutron Scattering



Coherent XS:



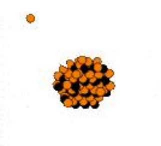
$$I_{trans}(t,\lambda) = I_{inc}(\lambda)\, e^{-\Sigma^* t}$$

Coherent XS~ Signal
Incoherent XS~ Background
Absorption XS~1/Intensity

Penetration depth~ 1/Sum of Scatt XS

Research Reactors

- Fission in Reactor Core
  - Moderated neutrons
  - Monochromators in guide
- "Constant Wavelength"
- Many Facilities in Europe:
  - ILL, Saclay (Fr), FRM-2 (G), Petten (NL), …
  - Generally low flux except ILL (and FRM-2)



Reactor Sources: Measurement of Strain

- Intensity as Function of diffraction angle $\Theta/2\Theta$

$$\boxed{\lambda = 2d\sin\theta}$$

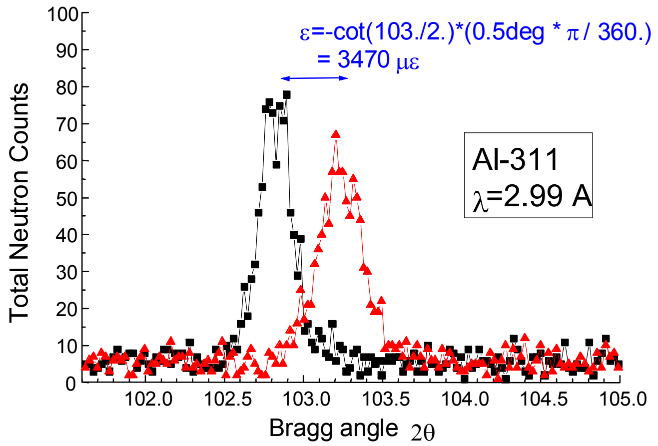
fixed

Strain: $\boxed{\varepsilon = -\cot(\theta_0)\,(\theta - \theta_0)}$

In units of *rad*

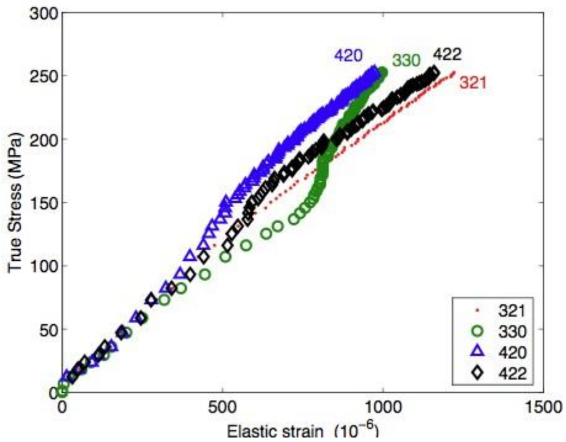
Single Wavelength at Reactor

Single-wavelength instrument: D1A at the ILL
New instrument at ILL: SALSA

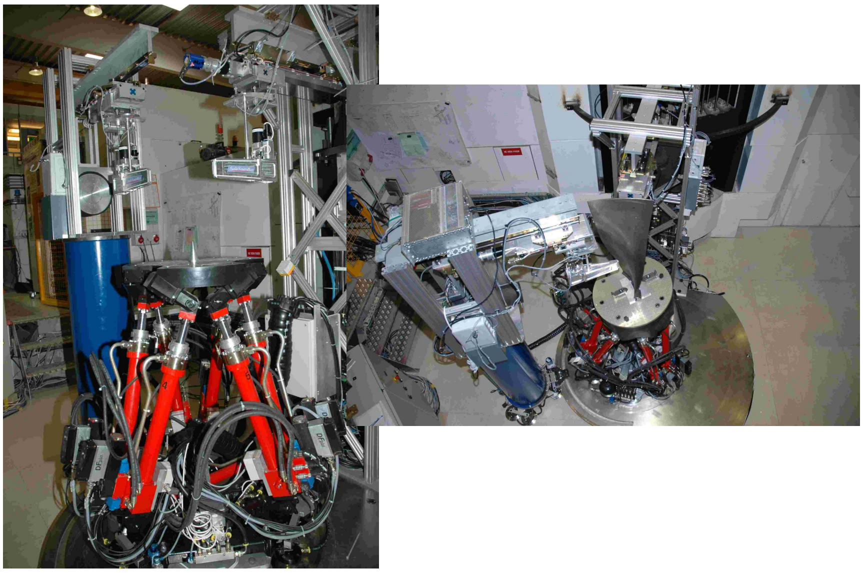


Which peak gives us the pure macrostress response ?

In-situ Loading on a neutron diffraction beam line

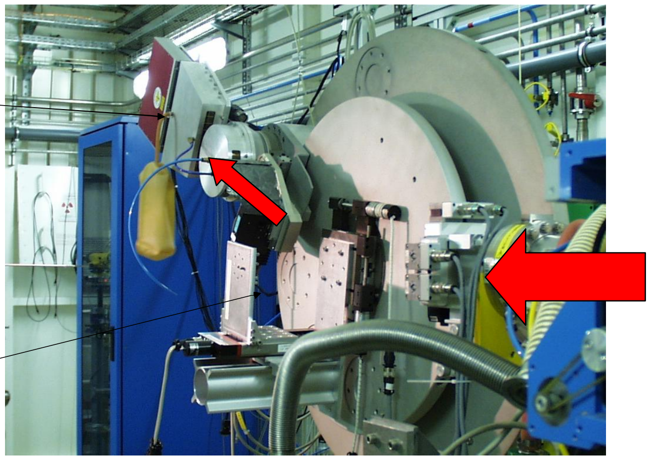


SALSA, ILL



Typical Diffractometer at Synchrotron (here ID31 at the ESRF)

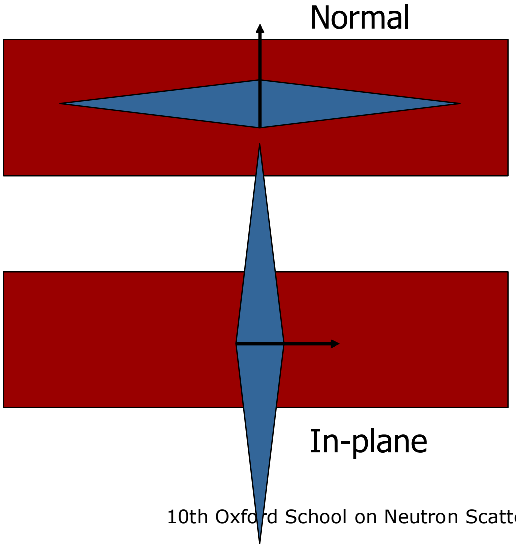
Detector



Sample Stage

Diamond gauge volume

Normal



In-plane

Two orthogonal strain components sample different material volume

Analyser crystal for partly filled Gauge volume necessary!

Time of flight method

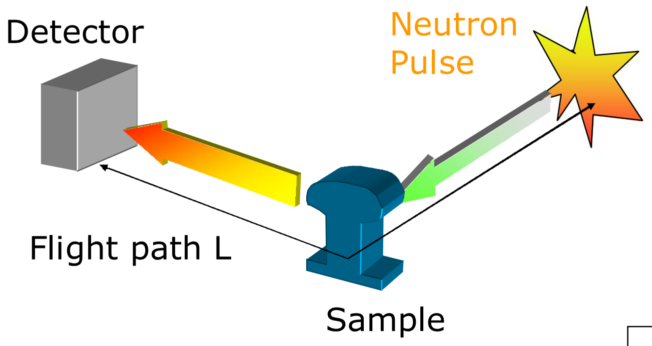
- Sharp pulse leaves source
- High energy neutrons (short λ) travel faster and arrive first, low energy (long λ) last

  $\lambda = ht/ml$

  where l is the path length and t time of flight
- a single stationary detector records whole diffraction spectrum as a function of time of flight
- neutrons travel at ~100m/s (speed of sound)

  $\lambda = 2d \sin \theta$ with θ fixed I.e. λ proportional to d

Spallation Sources: Time of Flight



$$\boxed{\lambda = \frac{h}{p} = \frac{h}{mv} = \frac{ht}{mL}}$$

$+$

$$\boxed{\lambda = 2d \sin \theta}$$

$=$

Time-of-Flight:

$$d = \frac{h}{2mL \sin \theta} t$$

Fast neutrons arrive earlier at detector!

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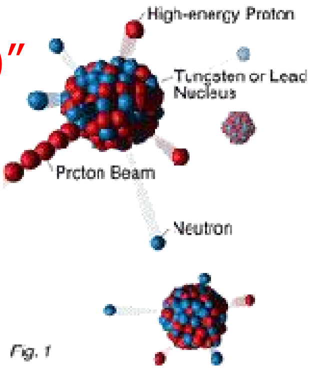
Spallation Sources

- Pulsed/Continuous Spallation Sources
- "Whole-Spectrum (TOF)"
- Facilities in Europe:
  - Pulsed: ISIS (UK)
  - POLDI (CH)
- More information

$$\lambda = 2d \sin \theta$$

fixed



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Spallation Sources: Measurement of Strain

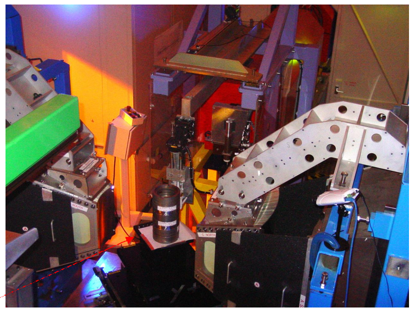
Strain: $\varepsilon = \frac{a - a_0}{a_0} = \frac{\lambda - \lambda_0}{\lambda_0} = \frac{t - t_0}{t_0}$

Cubic gauge volume !

Time-of-Flight:

$$\lambda = \frac{h}{2mL\sin\theta} t$$

Fixed



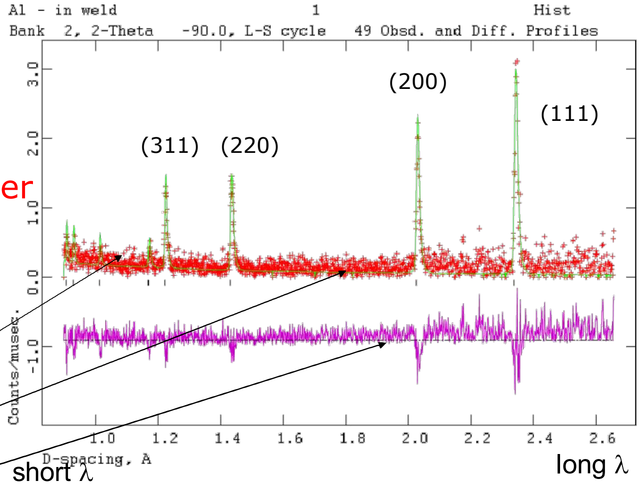
ENGIN-X at ISIS

Example of TOF: Rietveld Refinement

TOF: More Information!

Rietveld: Fewer Parameters!

$$d_{hkl} = \frac{a_0}{\sqrt{h^2 + k^2 + l^2}}$$



Data

Fitting

Difference

short λ

short times

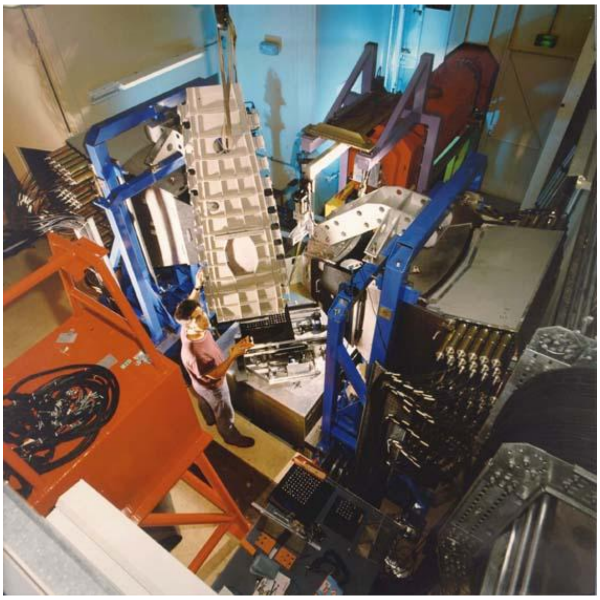
long λ

long times

ENGIN-X, ISIS

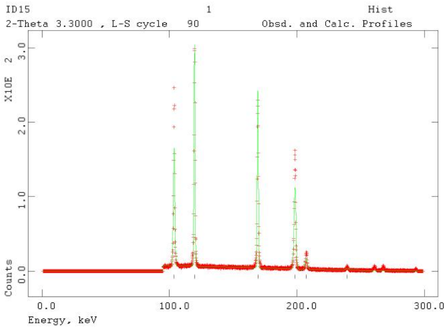
QuickTime™ and a
Microsoft Video 1 decompressor
are needed to see this picture.

ENGIN-X, ISIS



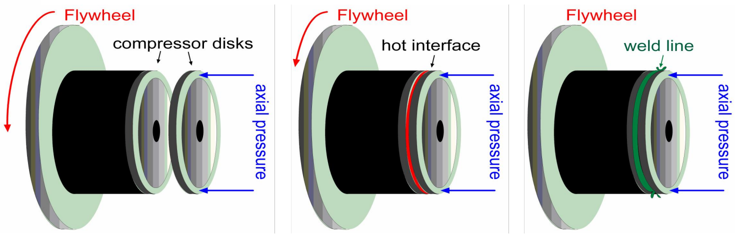
Energy Dispersive Synchrotron Diffraction

- Larger range of wavelengths available
- Energy/Strain Resolution up to 10E-5
- Higher penetration depth
- More elongated GV



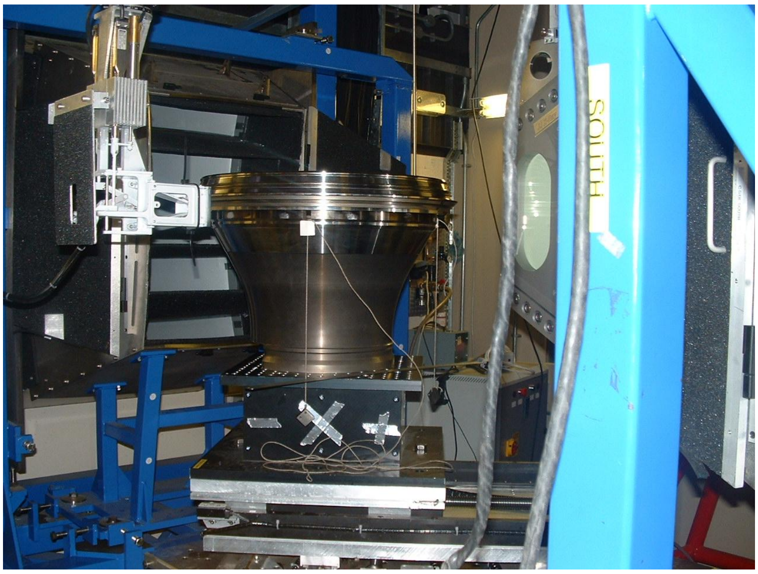
$$\boxed{\lambda = \frac{hc}{E} = \frac{12.39}{[keV]}} \Rightarrow \boxed{\varepsilon = \frac{E_0}{E} - 1}$$ Strain

Case Study: Inertia Friction Welding



Solid state joining of compressor, turbine discs and shafts

Case Study: Inertia Friction Welding

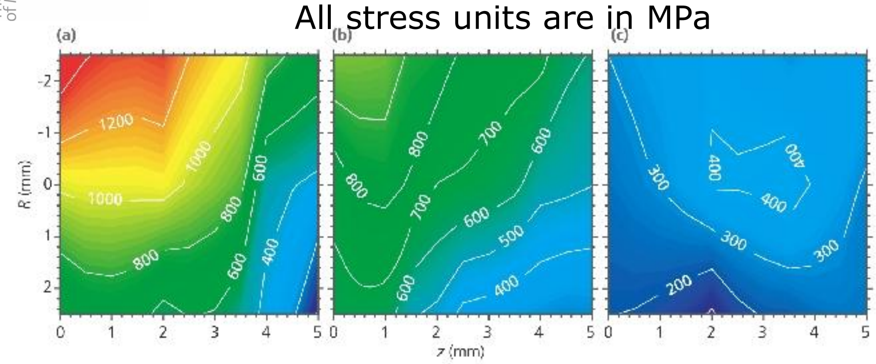


How would you measure such a sample ?

143mm diameter test inertia friction welds

Hoop stresses in IFW'd nickel-base superalloy

All stress units are in MPa

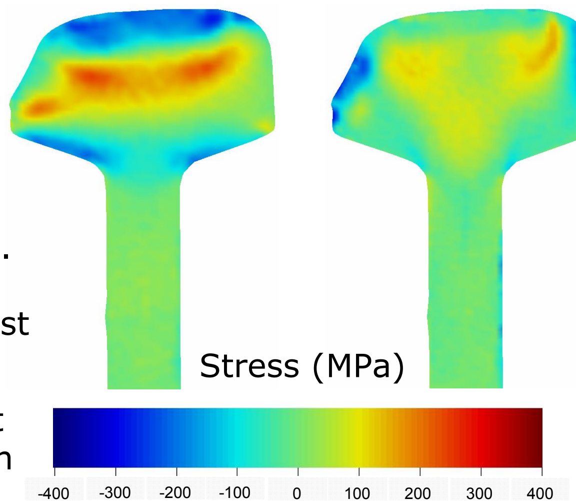


As-welded | Conventional PWHT | Modified PWHT

Residual stress measurements were used to develop a new PWHT

Railway Rails

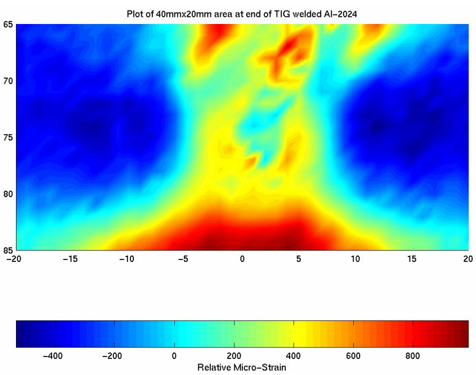
- Slices were cut from the rail to measure the horizontal and vertical stresses. Longitudinal stresses were lost
- Measurements were carried out by using neutron and synchrotron x-ray diffraction

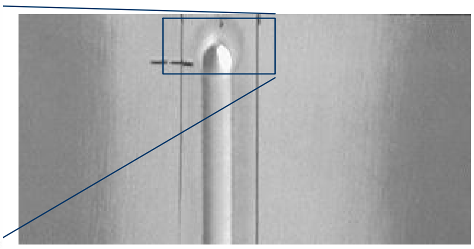


Case Study:
Strain Mapping of a TIG weld

2D Map of Residual Strain about the End of a TIG Weld at 100μm Resolution

This map include 20,00 measurements and took 8 hours to acquire

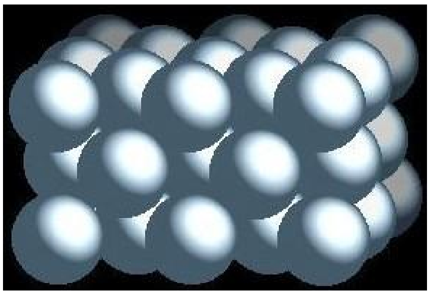




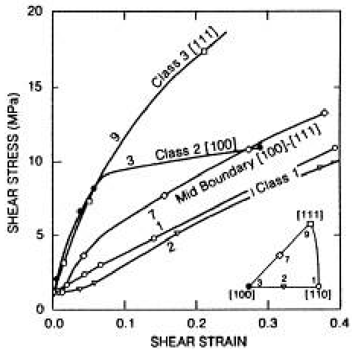
8mm wide Tig Weld in Aluminium Plate

From Engineering to Physical Metallurgy

Single Crystal Anisotropy



Al, fcc

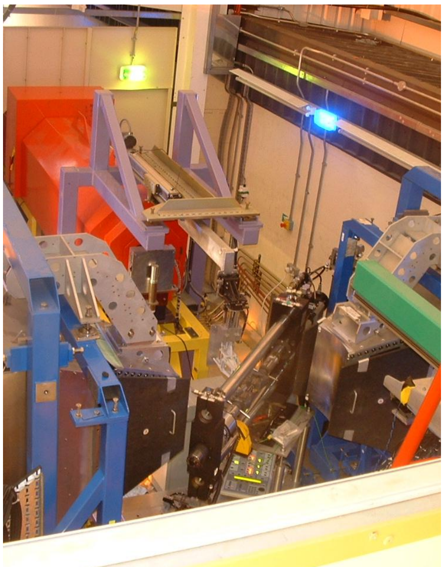


Single Crystal deformation

Deformation heterogeneity

- Polycrystalline deformation is heterogeneous
- Single crystal elastic and plastic anisotropy
- Grain incompatibility during deformation results in intergranular stresses

In-situ loading experiments



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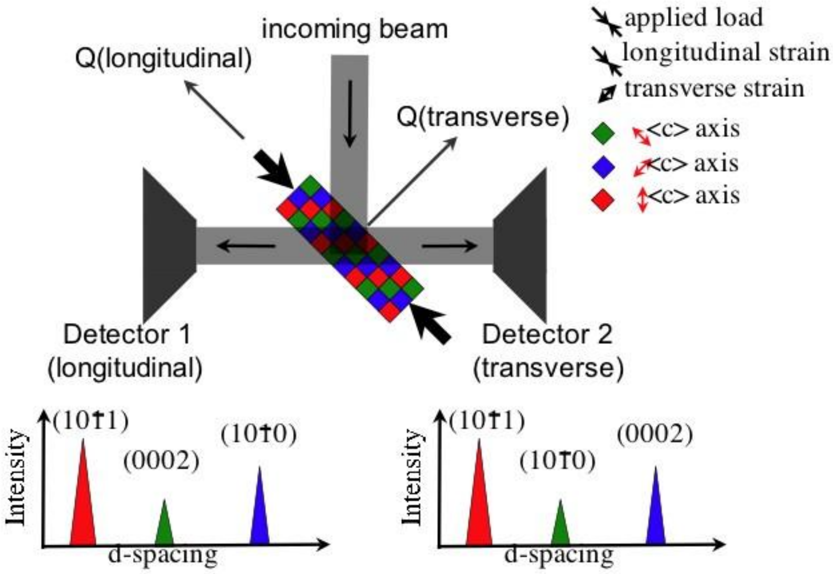
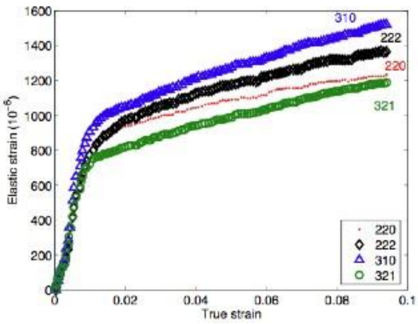
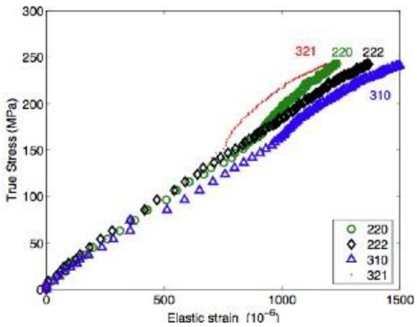
Principle of in-situ loading



In-situ straining

Elastic strain/elastic + plastic strain   Stress/elastic strain

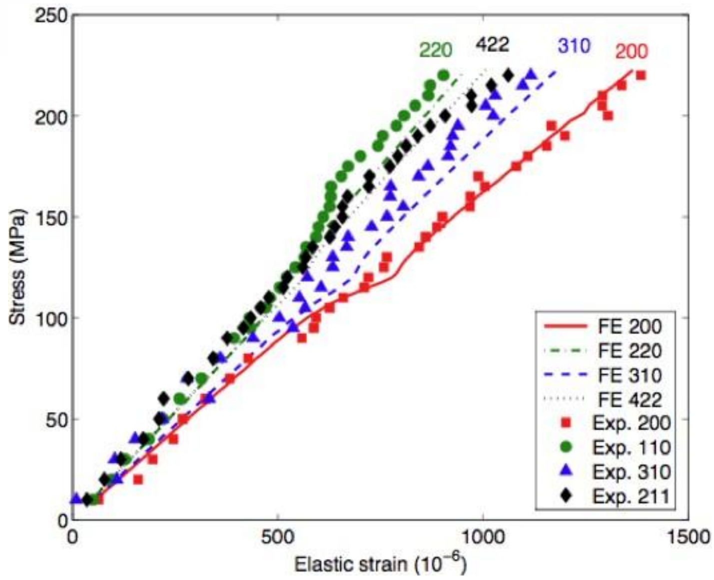




Modelling deformation

- Micromechanics
  - Dislocations, particles, grain boundaries (grain size), interstitial atoms
- Continuum mechanics:
  - Stresses and strains
  - Intergranular stresses
- Polycrystal plasticity
  - Mean filed methods, i.e. every grain has the same matrix
  - Finite element methods
    - Each grain has a characteristic neighbourhood
    - Predict maximum and minimum stresses ?

Plasticity Modelling (CPFEM)



Attempted General Guidelines: Neutrons

Neutrons:
•Non-destructive, full stress analysis because of cubic Gauge Volume (think three directions)
•Good penetration depth due to neutrality
•Big bulky sample with low stress gradients
•Reasonable spatial resolution independent of atomic number
•Steels, aluminium, nickel, copper zinc or related
•Sample in harsh environment: furnace, cryo. etc.
•Phase analysis with Rietveld analysis

Not-so good: near surface or thin materials, titanium, boron cadmium, fast, high-spatial resolution, high instrumental resolution, hydrogenous materials

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Attempted General Guidelines: Synchrotrons

Synchrotrons:

•Non-destructive, fast strain mapping, mostly single peak

•Light alloys (small atomic number)

•High spatial resolution aluminium-titanium (think microns)

•High instrumental resolution (small peak width)

•Near surface measurement because of analyser crystal

•Bulk materials / larger atomic number with energy-dispersive method

•Polymers

Not so good at: Steels and higher, big bulky samples, harsh environments, diamond shaped GV