A 'linear' elastic theory for hydrogels ITG LUNCH - 18TH MAY 2021

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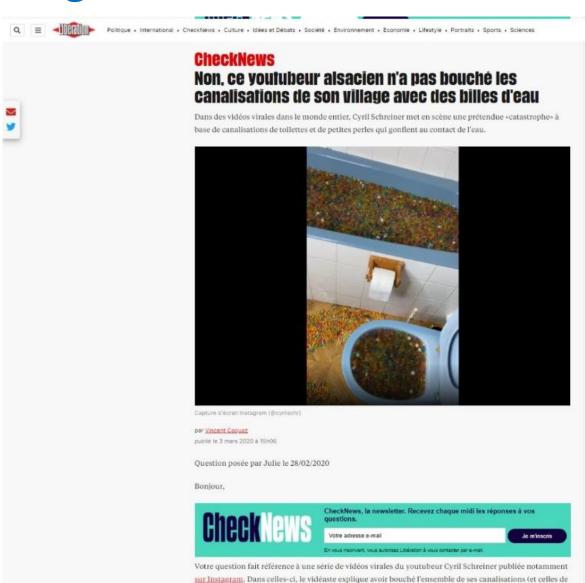


Hydrogels

- Hydrophilic polymer scaffold takes on water and thus can expand by >500% in volume from when dry.
- Applications include children's toys, absorbent nappies, contact lenses, wound dressings, water storage gels for pot plants.



Hydrogels



ses voisins) à cause de «perles d'eau», des petites billes qui gonflent une fois immergées dans de l'eau.



Pourquoi l'armée israélienne a-t-elle annoncé une attaque terrestre à Gaza qui n'a pas eu lieu?

14 mai 2021

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ecotorus 14 mai 2021

Que sait-on de cette vidéo d'un Arabe israélien lynché ?

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Hydrogels

- Hydrophilic polymer scaffold takes on water and thus can expand by >500% in volume from when dry.
- Applications include children's toys, absorbent nappies, contact lenses, wound dressings, water storage gels for pot plants.
- Also occur naturally cellulose, for example, is sufficiently hydrophilic to form hydrogels (with possible applications for bordered pits in plant xylem?)



 Usually studied as a 'bulk material' – solid and liquid phases are lumped together into one.

Why bother? The big picture

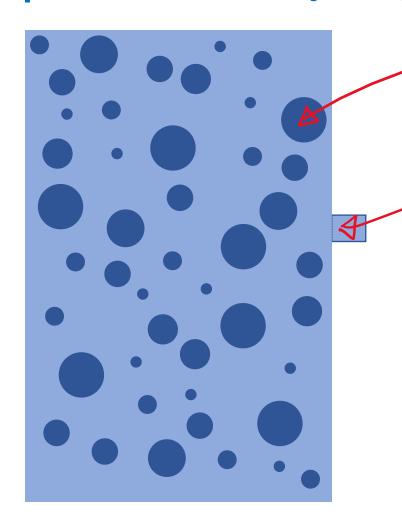
- It's clear that the properties of hydrogels arise from a number of different sources:
 - They exhibit elastic behaviour not just a property of the polymer scaffold!
 - The polymer chains are hydrophilic osmotic effects draw in water from the surrounds (it's this that leads to swelling).
 - Water can flow through the gel as is the case in bordered pits of trees.
- All of this suggests treating them as a poroelastic medium, in the same way that one
 would soils or other deformable porous media.

Why bother? The big picture

- Currently there are two main ways to model the poroelastic behaviour of hydrogels:
 - Nonlinear models (Gaussian-chain + Flory-Huggins) [Bertrand et al.] Gel treated as a single phase, with the elastic and osmotic effects from the work-energy function; describes the gel chemically but analytically intractable and only suitable for numerical computation.
 - Linear models [Doi, Biot, Tanaka-Hocker-Benedek etc.]

 Describe stresses on the gel using linear-elastic theory. Mathematically very simple but only suitable for small strains (~10% maximum according to Landau & Lifshitz).
- So neither is ideal for swelling and drying!

Pressure in hydrogels



Polymer fraction

Denoted ϕ - typically around 10^{-2}

Pervadic pressure *p*

The pressure of the liquid phase of the colloid as measured by a manometer separated from the system by a partially-permeable membrane.

Bulk (thermodynamic) pressure *P*

The force exerted per unit area by the whole system.

Osmotic pressure Π

The difference between these two: $P = p + \Pi$

Strains and deformation

• In gel swelling, expect *deviatoric* strains to be small but *isotropic* strains can be large. Lagrangian reference state $\phi = \phi_0$ everywhere, take $X \to X + \xi = x$, then

$$\underline{\underline{F}} = \nabla_{X} x$$
 with $\det \underline{\underline{F}} = \frac{\delta V}{\delta V_0} = \frac{\phi_0}{\phi}$

• Let $\underline{\underline{F}} = \mathcal{F}\underline{\underline{\mathbb{I}}} + \underline{\underline{f}}$ with the deviatoric part small. Then

$$\det \underline{\underline{F}} = \mathcal{F}^n + \mathcal{O}\left(\underline{\underline{f}^2}\right) \Rightarrow \mathcal{F} = \left(\frac{\phi_0}{\phi}\right)^{1/n}$$

To recast in an Eulerian point of view, define the Cauchy strain tensor

$$\underline{\underline{e}} = \frac{1}{2} (\nabla_{x} \xi + (\nabla_{x} \xi)^{T}) = \underline{\mathbb{I}} - \frac{1}{2} (\underline{\underline{F}^{-1}} + \underline{\underline{F}^{-T}})$$

Strains and deformation

Taylor-expand to first order, with

$$\underline{\underline{\boldsymbol{F}}^{-1}} = \frac{1}{\mathcal{F}} \underline{\underline{\mathbb{I}}} - \frac{1}{2\mathcal{F}^2} \left(\underline{\underline{\boldsymbol{f}}} + \underline{\underline{\boldsymbol{f}}^T}\right) + \mathcal{O}\left(\underline{\underline{\boldsymbol{f}}^2}\right)$$

$$\Rightarrow \underline{\underline{e}} = \left[1 - \left(\frac{\phi}{\phi_0}\right)^{1/n}\right] \underline{\underline{\mathbb{I}}} + \underline{\underline{\epsilon}}$$

Compare with linear limit

In linear elasticity,

$$\underline{\underline{e}} = -\frac{\phi - \phi_0}{n\phi_0} \underline{\mathbb{I}} + \underline{\underline{\epsilon}}$$

Deviatoric strain

Assumed small (to check later). Equal to $\frac{1}{2\mathcal{F}^2} \left(\underline{\underline{f}} + \underline{\underline{f}^T} \right)$

Which agrees with this in the $\phi \rightarrow \phi_0$ limit.

Gel dynamics

 The polymer and water phases are individually incompressible and thus we can use a standard mass conservation approach to describe dynamics of gel constituents:

$$\frac{\partial \phi}{\partial t} + \nabla \cdot (\phi u_p) = 0$$
 and $\frac{\partial (1-\phi)}{\partial t} + \nabla \cdot [(1-\phi)u_l] = 0$

- Let the Darcy (interstitial) velocity be $(1-\phi)ig(u_l-u_pig)=u$. Then $\nabla\cdot u+\nabla\cdot u_p=0$.
- Combine with Darcy's law to find that

$$\nabla \cdot \boldsymbol{u_p} = \frac{\partial (\nabla \cdot \boldsymbol{\xi})}{\partial t} = \frac{\phi^{(1-n)/n}}{\phi_0^{1/n}} \frac{\partial \phi}{\partial t} = \frac{1}{\mu_l} \nabla \cdot [k(\phi) \nabla p]$$

• This relates polymer fractions to pervadic pressure gradients.

Towards a constitutive relation

- We now wish to relate gradients in p to gel properties ξ and ϕ .
- Separate out the stress tensor into a part due to the solid and one due to the liquid:

$$\underline{\underline{\boldsymbol{\sigma}}} = \phi \underline{\underline{\boldsymbol{\sigma}}}_{s} + (1 - \phi) \underline{\underline{\boldsymbol{\sigma}}}_{l}$$

• Assume a Newtonian constitutive relation for the fluid with viscosity μ_l and follow Terzaghi in introducing an effective stress.

$$\underline{\underline{\boldsymbol{\sigma}}} = \phi(\underline{\underline{\boldsymbol{\sigma}_s}} + p\underline{\underline{\mathbb{I}}}) + 2(1 - \phi)\mu_l\underline{\underline{\boldsymbol{\varepsilon}}} - p\underline{\underline{\mathbb{I}}}$$

• Neglect the viscous stresses and seek a form for the effective stress $\underline{\underline{\sigma_e}} = \phi(\underline{\underline{\sigma_s}} + p\underline{\underline{\mathbb{I}}}).$

Linear-elastic-nonlinear-swelling approach

- Linear elasticity assumes $\underline{\underline{\sigma}_{e}} = \underline{\underline{C}} : \underline{\underline{e}}$ for some constant tensor $\underline{\underline{C}}$, then by isotropy $\underline{\underline{\sigma}_{e}} = \left(K \operatorname{tr} \underline{\underline{e}} \right) \underline{\underline{\mathbb{I}}} + 2\mu_{s} \underline{\underline{\epsilon}}$
- We allow $\underline{\underline{C}}$, and thus the material properties, to depend on ϕ , and thus, by a similar argument, have

$$\underline{\underline{\sigma_e}} = f(\phi)\underline{\underline{\mathbb{I}}} + 2\mu_s(\phi)\underline{\underline{\epsilon}} = -K(\phi)\frac{\phi - \phi_0}{\phi_0}\underline{\underline{\mathbb{I}}} + 2\mu_s(\phi)\underline{\underline{\epsilon}}$$

$$\underline{\underline{\sigma}} = -\left[p + K(\phi)\frac{\phi - \phi_0}{\phi_0}\right] \underline{\underline{\mathbb{I}}} + 2\mu_s(\phi)\underline{\underline{\epsilon}}$$
 Shear modulus

Osmotic modulus

Linear-elastic-nonlinear-swelling approach

• Material properties are described by just four parameters $(K, \mu_s, \phi_0 \text{ and } k)$ with governing equation and constitutive relation

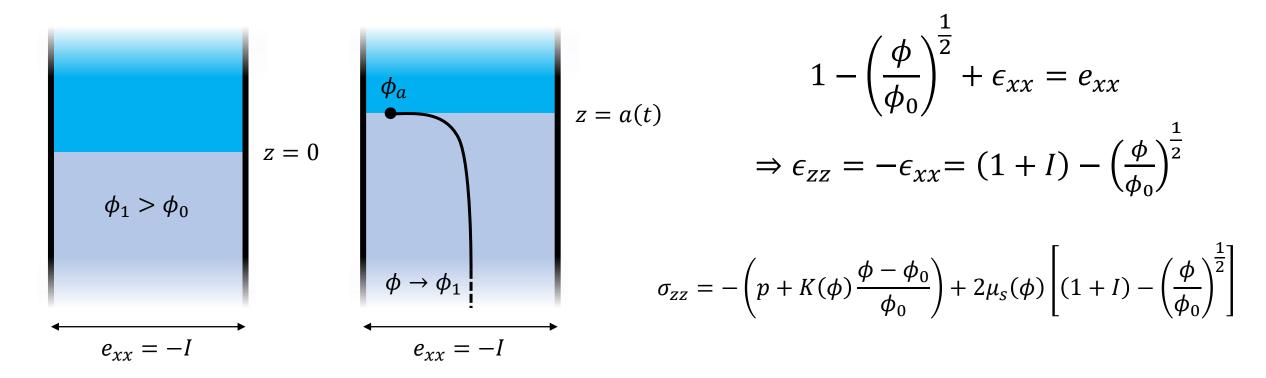
$$\frac{\partial \phi}{\partial t} = \frac{\phi_0^{1/n} \phi^{(n-1)/n}}{\mu_l} \nabla \cdot [k(\phi) \nabla p] \quad \text{and} \quad \underline{\underline{\boldsymbol{\sigma}}} = -\left[p + K(\phi) \frac{\phi - \phi_0}{\phi_0}\right] \underline{\mathbb{I}} + 2\mu_s(\phi) \underline{\underline{\boldsymbol{\epsilon}}}.$$

• Still need to justify neglecting viscous terms. Scaling argument:

$$\frac{\text{viscous}}{\text{elastic}} \sim \frac{(1-\phi_0)\mu_l u_l}{\mu_s u_p t_{el}} \sim \frac{\phi_0 \mu_l}{\mu_s t_{el}} \sim \frac{\phi_0 k_0 K}{\mu_s L^2} \ll 1$$

$$t_{el} \sim \frac{\mu_l L^2}{k_0 K}, \text{ letting } p \sim K$$

- Take partially-dried hydrogel confined horizontally, with uniform polymer fraction $\phi_1 > \phi_0$. The confinement means that the gel can't swell any more in the horizontal direction, but will swell vertically.
- The confinement also means we can express ϵ_{zz} in terms of I and ϕ .



$$\sigma_{zz} = -\left(p + K\frac{\phi - \phi_0}{\phi_0}\right) + 2\mu_s \left[(1+I) - \left(\frac{\phi}{\phi_0}\right)^{\frac{1}{2}} \right]$$

• The equation of equilibrium $\partial \sigma_{zz}/\partial z=0$ gives an expression for $\partial p/\partial z$. Substitute into polymer conservation law to get

$$\frac{\partial \phi}{\partial t} = \frac{\phi_0^{1/2} \phi^{1/2}}{\mu_l} \frac{\partial}{\partial z} \left[k(\phi) \left\{ \frac{K(\phi)}{\phi_0} + \frac{\phi - \phi_0}{\phi_0} \frac{\partial K}{\partial \phi} + \frac{\mu_s(\phi)}{\phi_0^{1/2} \phi^{1/2}} - 2 \left[(1+I) - \left(\frac{\phi}{\phi_0} \right)^{1/2} \right] \frac{\partial \mu_s}{\partial \phi} \right\} \frac{\partial \phi}{\partial z} \right]$$
(!)

"Diffusivity" increases for larger K, k, smaller μ_l

- Boundary condition at interface: $p = \sigma_{zz} = 0$ at z = a(t) this sets interfacial ϕ .
- Conservation of polymer sets $\int_{-\infty}^{a(t)} (\phi \phi_1) dz + \phi_1 a(t) = 0$.
- This is messy but powerful. Take K, μ_S , k_0 to be constant, and it looks altogether more familiar in the limit $\phi \to \phi_0$.

• In the "fully linear" (Biot) limit, we're left with a problem that can easily be solved analytically.

$$\frac{\partial \phi}{\partial t} = \frac{k_0(K + \mu_S)}{\mu_l} \frac{\partial^2 \phi}{\partial z^2}$$

$$\phi \to \phi_0(1 + 2I) \text{ as } z \to -\infty$$

$$\phi = \phi_0[1 + 2\mu_S I/(K + \mu_S)] \text{ at } z = a(t)$$

$$\dot{a} = -\frac{k_0(K + \mu_S)}{\mu_l} \frac{\partial \phi}{\partial z} \bigg|_{z=a(t)}$$

Far-field boundary condition

Assume isotropic swelling so $\underline{\underline{e}} = -I \underline{\mathbb{I}}$ and thus $\phi_1 = \phi_0(1 + 2I)$

Interfacial boundary condition

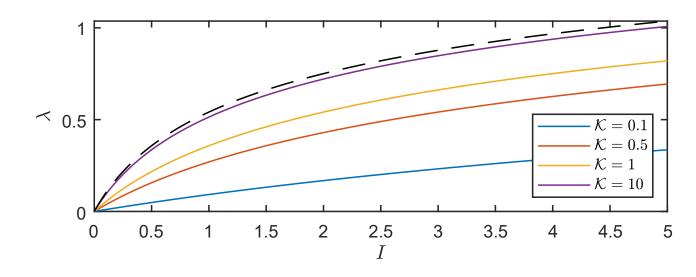
Interfacial polymer fraction comes from setting $\Pi(\phi) = 2\mu_s \epsilon_{zz}$

$$\phi = \phi_0 \left[1 + 2I \left(1 - \frac{\mathcal{K}}{1 + \mathcal{K}} \frac{\operatorname{erfc}(-\eta)}{\operatorname{erfc}(-\lambda)} \right) \right]$$
$$a(t) = 2\lambda \sqrt{Dt}$$

with

$$\mathcal{K} = K/\mu_S$$
, $D = \frac{k_0(\mu_S + K)}{\mu_l}$ and

$$\sqrt{\pi}\lambda e^{\lambda^2}\operatorname{erfc}(-\lambda) = \frac{2I\mathcal{K}}{1+\mathcal{K}} = \frac{\phi_1 - \phi_a}{\phi_0}$$



- Growth rate is dependent on the contrast between initial and final polymer fraction; more 'confinement' implies faster growth.
- Remark: the growing interface is linearly unstable here – instability can be analysed with Biot's approach.

Swelling of a hydrogel bead

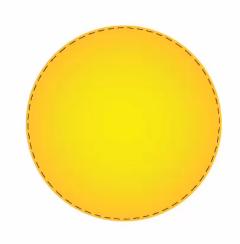
- Problem has been considered by many authors both in nonlinear [Bertrand et al.] and linear [Punter et al.] contexts.
- We want a post-hoc justification that the assumption of large isotropic strains but small deviatoric strains is valid.

Problem setup

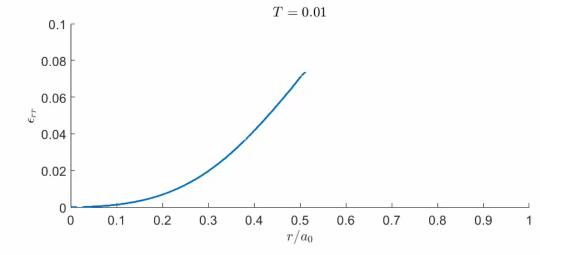
- Spherical bead with uniform polymer fraction $\phi_1 > \phi_0$ placed in bulk water.
- Force-free, so $\nabla \cdot \underline{\underline{\sigma}} = \mathbf{0}$ sets a relation between p and the other variables.
- Pervadic pressure and radial stress continuous across interface so p=0 and $\sigma_{rr}=0$ here as a boundary condition.
- Assume $k(\phi) = k_0$ and μ_s , K constants (for now).

Swelling of a hydrogel bead

$$\frac{K}{\mu_s}=0.025$$
 and $oldsymbol{\phi}=8oldsymbol{\phi}_0$ initially



$$\epsilon_{rr} = \frac{2}{3} \left(\frac{\partial \xi}{\partial r} - \frac{\xi}{r} \right) = 2 \left[1 - \frac{\xi}{r} - \left(\frac{\phi}{\phi_0} \right)^{1/3} \right] \ll 1?$$

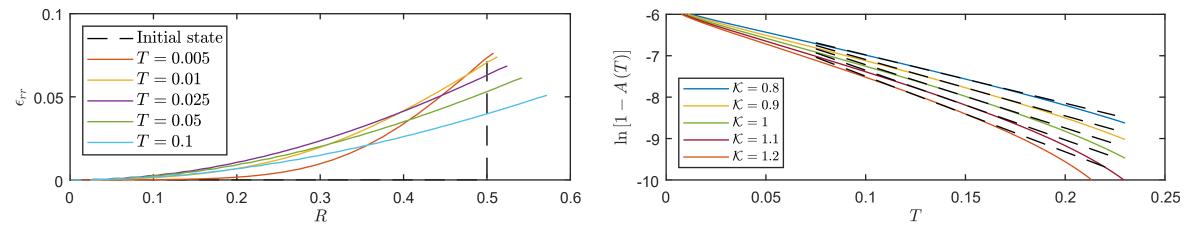


Boundary condition becomes

$$K\frac{\phi - \phi_0}{\phi_0} = 2\mu_s \epsilon_{rr}$$
on $r = a(t)$.

Swelling of a hydrogel bead

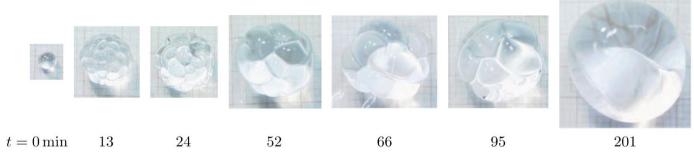
 Thus, can model swelling at very large strains but the transient state never has large deviatoric strains – which supports the initial assumptions of our model.



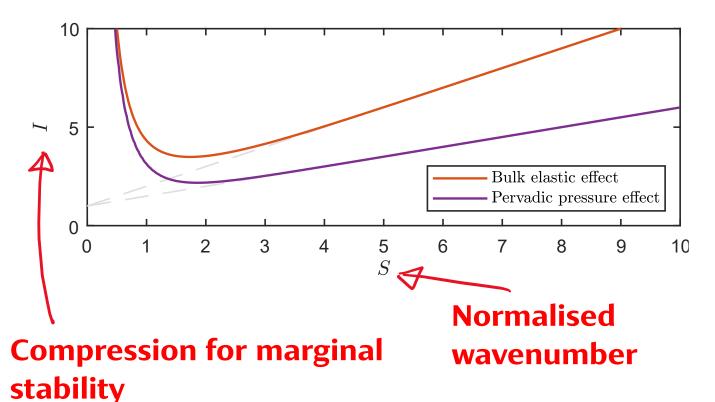
• This model also agrees well with existing linear models in the limit of small swelling, such as that of Punter *et al.* (as shown above).

Where next?

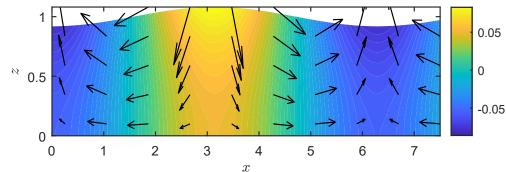
- Aim to investigate surface tension effects at curved interfaces such effects are well-documented in the literature.
- How does the surface tension arise; is it a discontinuity in elastic stress or a discontinuity in the pervadic pressure which creates the condition $\left[\underline{\boldsymbol{\sigma}}\cdot\boldsymbol{n}\right]^+=\gamma\kappa\boldsymbol{n}$ (ideally we want the latter, but how to check this?)
- First idea consider the aforementioned swelling instability for hydrogels do we get qualitatively different behaviour if surface tension is introduced differently?



Where next?



The value of *I* at which the system is marginally stable



Making the discontinuity one in pervadic pressure drives a destabilising interstitial flow through the hydrogel matrix which quantitatively, but not qualitatively, destabilises the situation.

Conclusions

- Have developed a theory which couples linear elasticity in deviatoric strains with nonlinear behaviour in isotropic (swelling) strains both in constitutive relations and dynamical equations.
- This has the advantages of being analytically tractable but also captures perhaps the most important behaviour of hydrogels – the fact they can swell greatly.
- Next steps involve considering surface tension effects can we quantify what the mechanism is behind this?
- Also seek constitutive relations for gels, using rheometry.