

Solution of the Kohn-Sham equations

We now have an *ab-initio*, approximate DFT which can be used in practice to compute properties of molecules and condensed systems.

The basic variable: electronic charge density

$$\rho = \sum_{i=1}^N |\psi_i(\vec{r})|^2 \quad \text{with} \quad \langle \psi_i | \psi_j \rangle = \delta_{ij}$$

The total energy in terms of the charge density and the single particle orbitals

$$E_{KS} = -\frac{1}{2} \sum_{i=1}^N \int \psi_i^*(\vec{r}) \nabla^2 \psi_i(\vec{r}) d\vec{r} + \frac{1}{2} \int d\vec{r} d\vec{r}' \frac{\rho(\vec{r}) \rho(\vec{r}')}{|\vec{r} - \vec{r}'|} + \int d\vec{r} \rho(\vec{r}) \epsilon_{xc}(\rho) + \int d\vec{r} V_{\text{ext}}(\vec{r}) \rho(\vec{r})$$

The Kohn-Sham equations: set of N non linear differential equations for the single particles orbitals; the potential V_{KS} depends 'self-consistently' upon the charge density

$$\left(-\frac{1}{2} \nabla^2 + V_{KS}[\rho(\vec{r})] \right) \psi_i(\vec{r}) = \epsilon_i \psi_i(\vec{r}) \quad \text{with} \quad V_{KS} = \frac{\delta E_{KS}}{\delta \rho(\vec{r})}$$

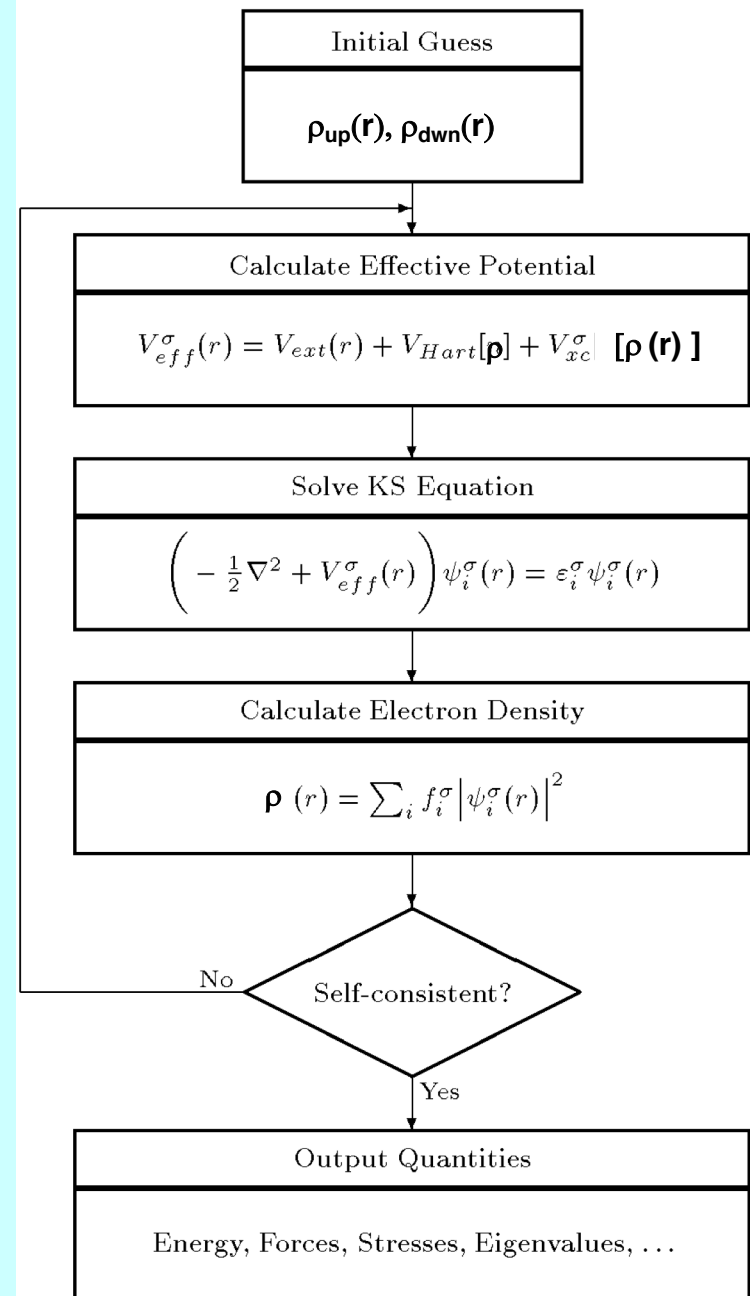
Approximations (e.g. LDA, GGA, etc.) for the exchange correlation energy density

ϵ_{xc} and $E_{xc}[\rho]$ are in general unknown and one needs to resort to approximations

Solving KS equations

- Choose **approximation for exchange and correlation potential**
- Choose **atomic configuration**, i.e. positions of the nuclei \mathbf{R}_I
- Choose:
 - **how to treat atomic cores** (all-electron, frozen core, pseudopotentials)
 - **basis set** to expand single particle orbitals
- Start self-consistent cycle by “**guessing**” a **set of single particle orbitals** (i.e. the expansion coefficients in terms of a given basis set) and thus an input charge density ρ .
- Solve KS **eigenvalue problem** in an **iterative manner**.

Self-Consistent Kohn-Sham Equations



Compare with solution of Hartree-Fock equations

How do we solve the Hartree-Fock equations, i.e. an eigenvalue problem where the operator depends on the solutions? **Using an self-consistent iterative procedure**

Guess an initial Slater determinant: $\psi \approx \psi \equiv |\psi_1^{in} \dots \psi_N^{in}|$

Constructs the Coulomb and exchange potentials using the orbitals from the guessed Slater determinant $J[\{\psi_i^{in}\}] ; K[\{\psi_i^{in}\}]$

Construct the Fock operator and **solve the eigenvalue problem** (e.g. my matrix diagonalization, after expanding the single particle orbitals in a suitable **basis set**, for example a Gaussian basis set):

$$\hat{F}[\{\psi_i^{in}\}; \vec{r}] \psi_m^{out} = \epsilon_m^{out} \psi_m^{out}$$

$$\psi^{in} \approx \psi^{out}$$

yes

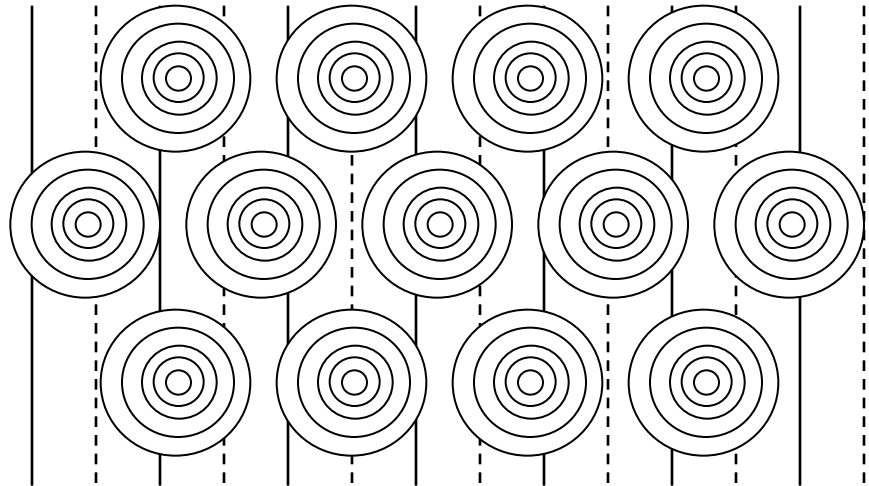
NO

$$\psi \approx \alpha \psi^{in} + \beta \psi^{out}$$

Self-consistency attained; Compute total energies and orbital energies

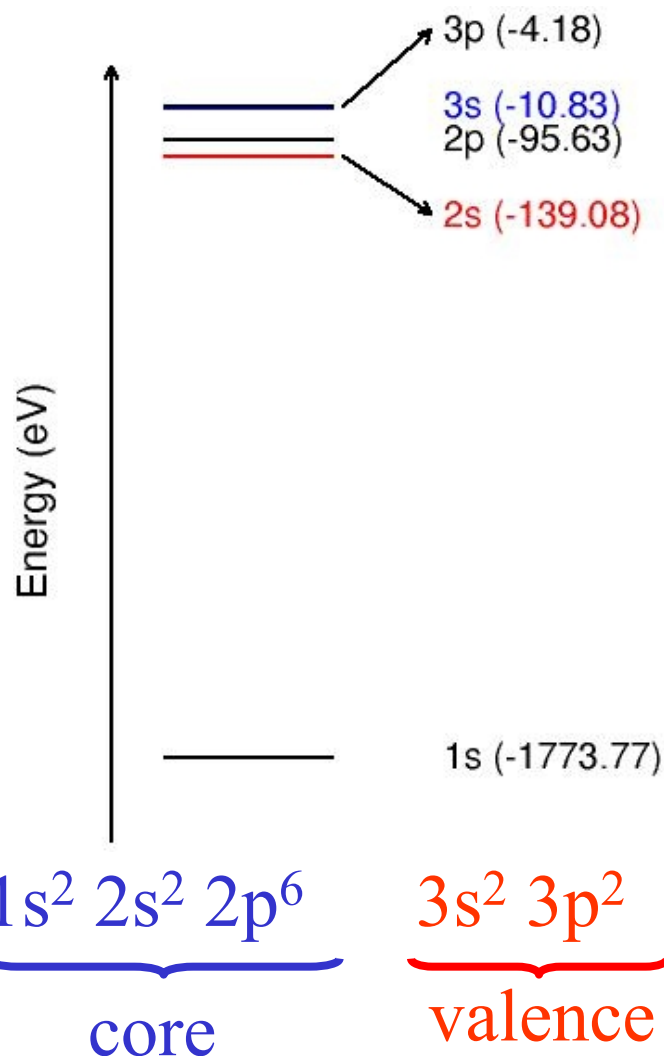
Representation of atomic cores

- **All-electron (AE) methods:** all electrons in the system are treated explicitly; some of them may be considered “frozen” in the configuration they have in the atom
- **Pseudopotential (PP) method:** only **valence electrons** are treated explicitly
 - **Core states – strongly bound to nuclei – atomic-like**
 - **Valence states (treated explicitly) –** they undergo changes when going from an atom to a molecule or a condensed system; the way they change determine the bonding, electronic and optical properties, magnetism,



Pseudopotentials and pseudo-wavefunctions (I)

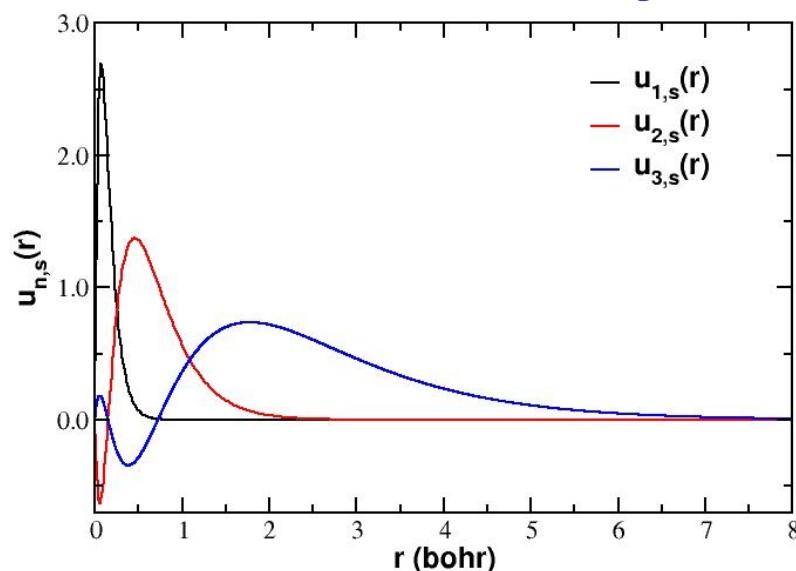
Atomic Si



Core electrons...

highly localized
very low energy
compared to valence
states

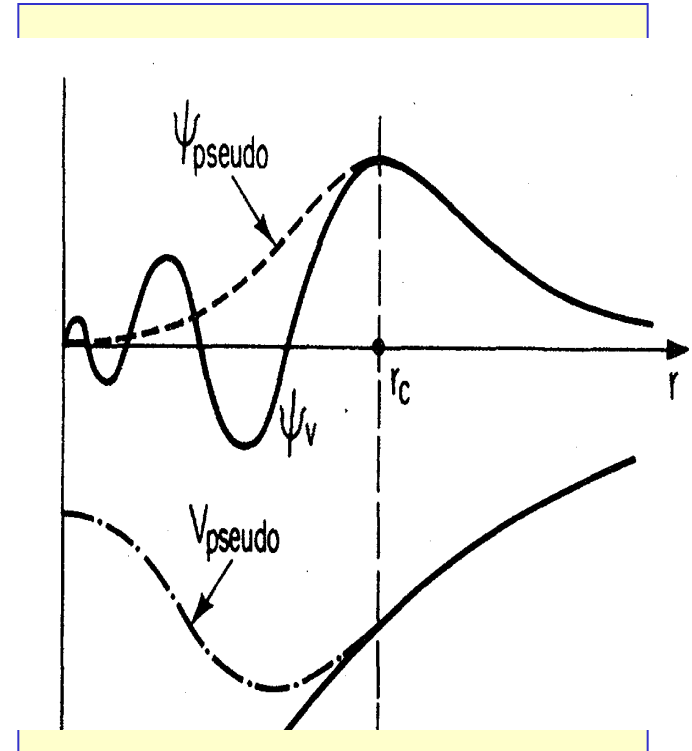
... are chemically inert



Valence wave functions must be orthogonal to the core wave functions

Pseudopotentials and pseudo-wavefunctions (II)

- Idea: construct ‘pseudo-atom’ which has the valence states of the real atom as its lowest electronic states
- Preserves scattering properties and total energy *differences* for the real atom
- Removal of orbital nodes makes plane-wave expansion feasible
- Possible limitations: Can the pseudo-atom correctly describe the bonding in different environments ?
→ transferability of pseudo-potentials in different chemical environments needs to be checked.



Transferability of pseudopotentials: atoms

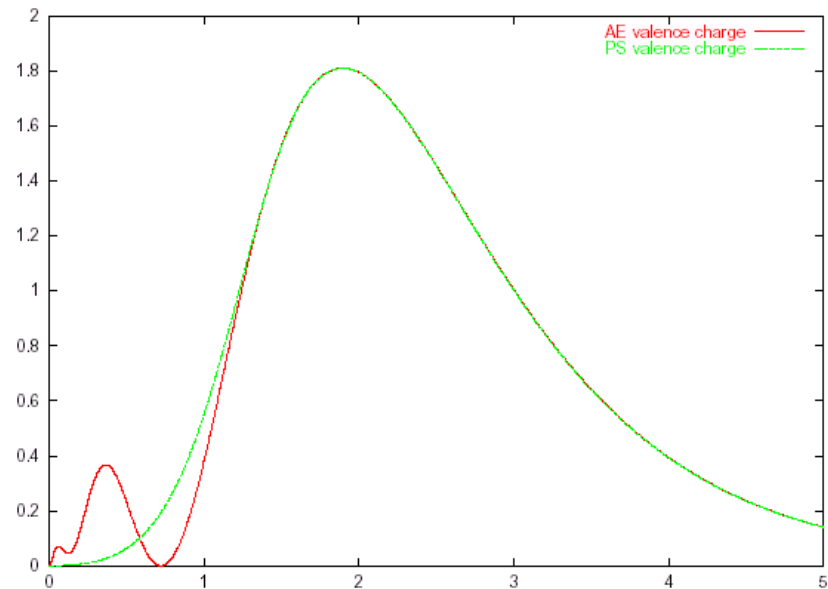
- Compute the energy of two different configurations with AE and PP methods for the chosen atom
- Compute the difference in energy using both AE and PP methods
- For the pseudopotential to be transferable: $\Delta E_{AE} = \Delta E_{PP}$

Example of configurations for the Si atom:

- $3s^2 3p^2$ (reference)
- $3s^2 3p^1 3d^1$
- $3s^1 3p^3$
- $3s^1 3p^2 3d^1$
- $3s^0 3p^3 3d^1$

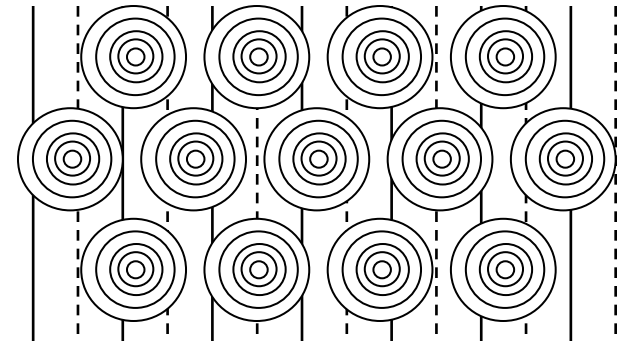
Pseudopotentials and pseudo-wavefunctions (III)

Comparison between all electron and pseudo-charge density



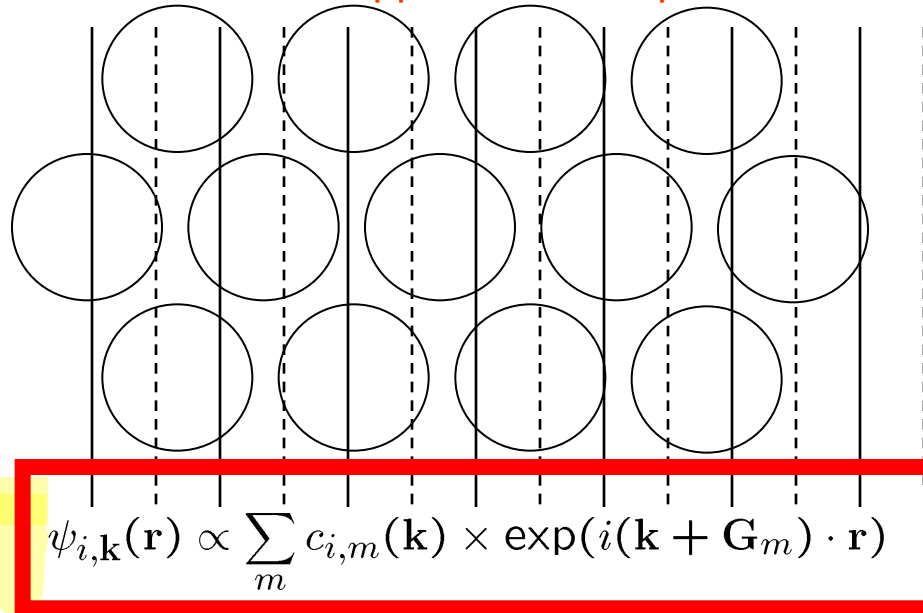
Four choices for single particle orbital representation in modern DFT calculations

- **Plane waves**
 - The simplicity of Fourier Expansions
 - The speed of Fast Fourier Transforms
 - Requires smooth pseudopotentials
- **Localized orbitals**
 - The intuitive appeal of atomic-like states
 - Simplest interpretation in tight-binding form
 - Gaussian basis widely used for finite systems (within all quantum in chemistry)
- **Augmented methods**
 - “Best of both (plane waves and localized orbital) worlds” – also most demanding
 - Requires matching technique to match functions inside and outside spheres.
- **Real Space representation of wavefunctions, charge densities and potentials** (not as popular as the other three methods)



Plane wave basis sets

- The most general and unbiased approach: no dependence on nuclei positions



$$\sum_{m'} H_{m,m'}(\mathbf{k}) c_{i,m'}(\mathbf{k}) = \varepsilon_i(\mathbf{k}) c_{i,m}(\mathbf{k})$$

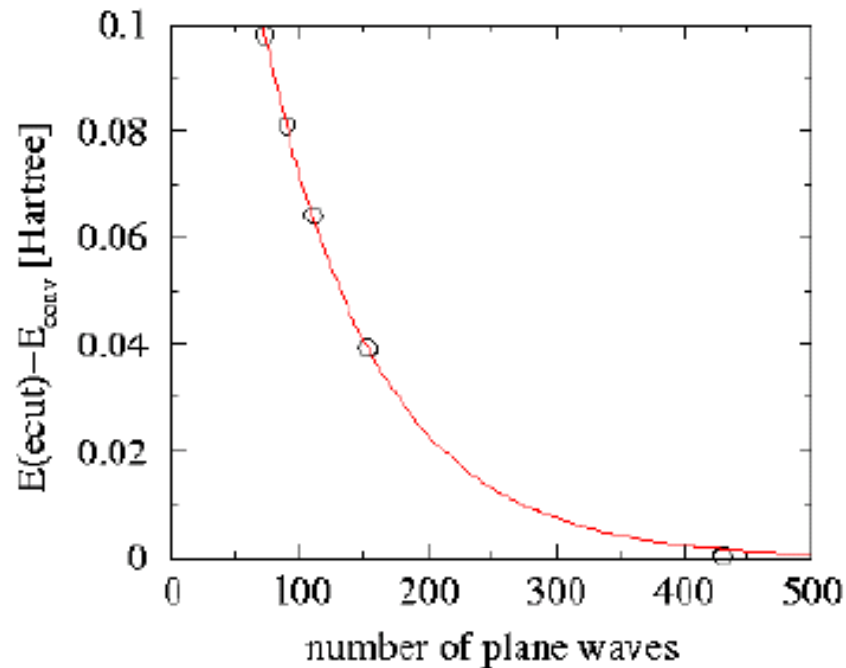
$$H_{m,m'}(\mathbf{k}) = \frac{\hbar^2}{2m_e} |\mathbf{k} + \mathbf{G}_m|^2 \delta_{m,m'} + V_{eff}(\mathbf{G}_m - \mathbf{G}_{m'}).$$

- The problem is the atoms! High Fourier components. Solution: use pseudopotentials.

Convergence as a function of # of PWs (or PW energy cutoff)

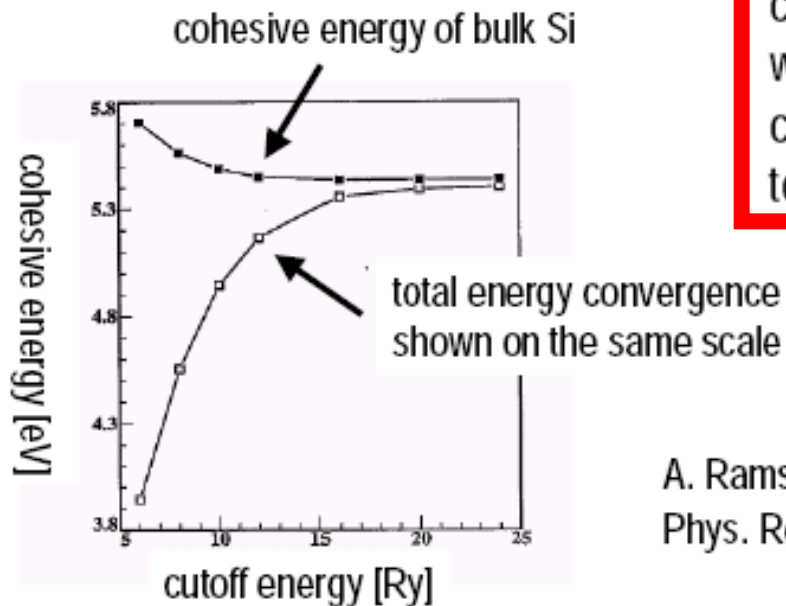
Example 1:

absolute convergence
for GaAs, LDA, Hamann
pseudopotential,
data points at
 $E_{\text{cut}} = 7, 8, 10, 20 \text{ Ry}$



Convergence as a function of # of PWs (or PW energy cutoff)

Example 2: Si

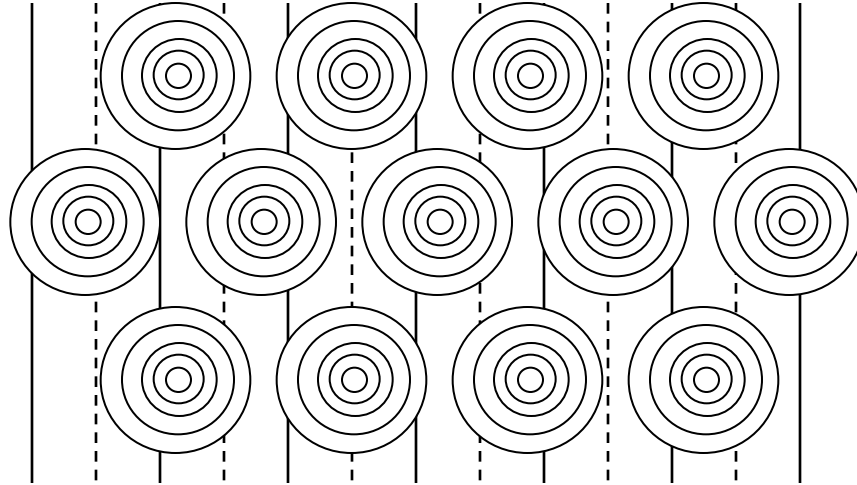


relative energies
converge much faster
with the plane wave
cutoff than the absolute
total energy !

A. Ramstad, G. Brocks, and P. J. Kelly,
Phys. Rev. B **51**, 14504 (1995).

Augmented plane waves (I)

- (L)APW method



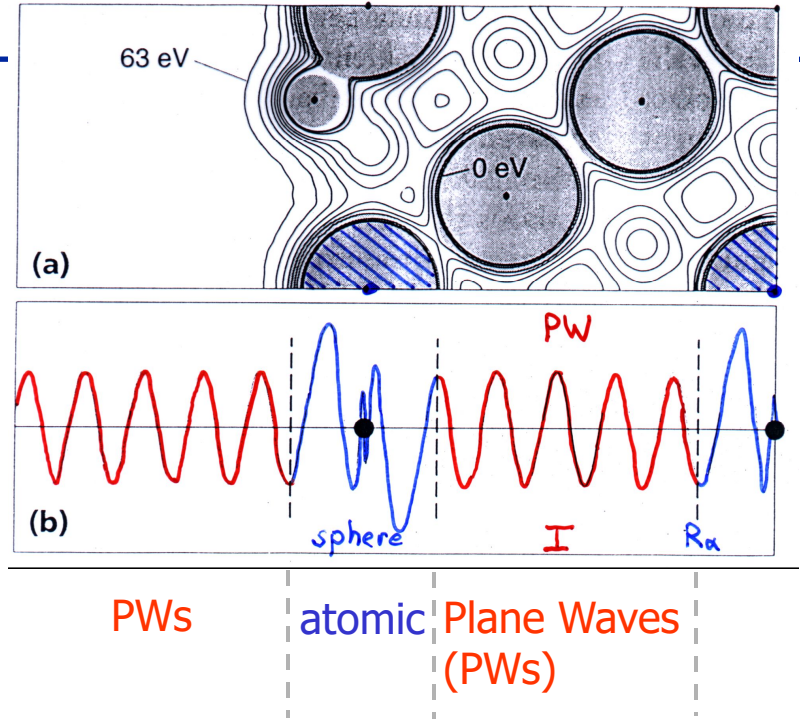
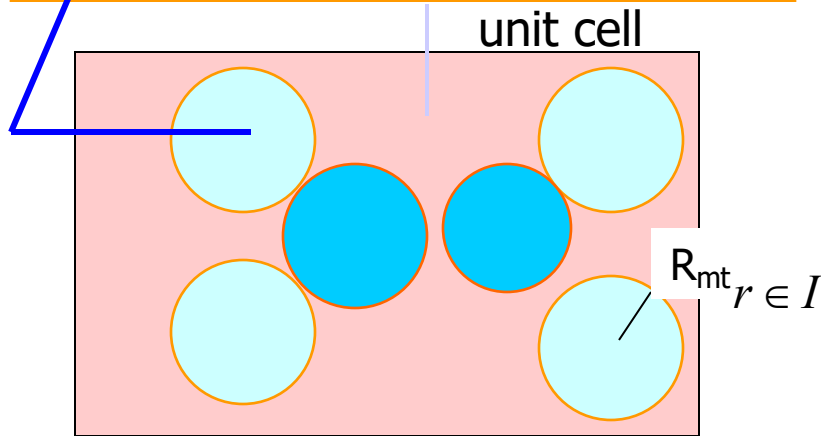
- **Augmentation:** represent the wave function inside given spheres centered on atoms in spherical harmonics, and as plane wave outside.
 - “Best of both worlds”
 - Requires matching inside and outside functions
 - Most general form – can approach arbitrarily precision as it does not require the use of pseudopotentials.

Augmented plane waves (II)

The unit cell is partitioned into:

atomic spheres

Interstitial region



Basis set:

PW: $e^{i(\vec{k} + \vec{K}) \cdot \vec{r}}$

Atomic partial waves

$$\sum_{\ell m} A_{\ell m}^K u_{\ell}(r', \varepsilon) Y_{\ell m}(\hat{r}')$$

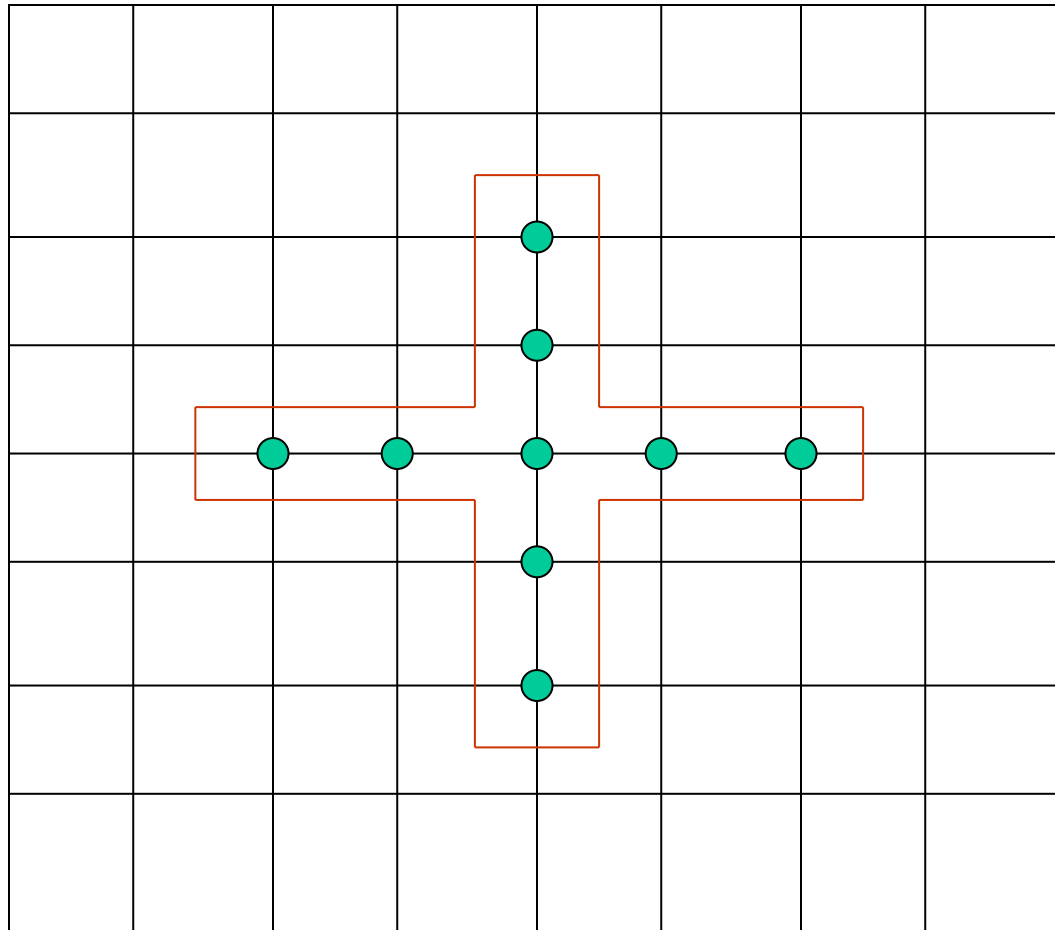
join

$u_{\ell}(r, \varepsilon)$ are the numerical solutions of the radial Schrödinger equation in a given spherical potential for a particular energy ε
 $A_{\ell m}^K$ coefficients for matching the PW

Finite-difference representation of KS eigenvalue problem:

$$\begin{aligned} & -\frac{1}{2h^2} \begin{bmatrix} -2 & 1 & 0 & \dots \\ 1 & -2 & 1 & \dots \\ 0 & 1 & -2 & \dots \\ \vdots & \vdots & \vdots & \ddots \end{bmatrix} \begin{bmatrix} \psi(x_1) \\ \cdot \\ \cdot \\ \cdot \\ \psi(x_N) \end{bmatrix} \\ & + \begin{bmatrix} V(x_1) & 0 & 0 & \dots \\ 0 & V(x_2) & 0 & \dots \\ 0 & 0 & V(x_3) & \dots \\ \vdots & \vdots & \vdots & \ddots \end{bmatrix} \begin{bmatrix} \psi(x_1) \\ \cdot \\ \cdot \\ \cdot \\ \psi(x_N) \end{bmatrix} \\ & = \lambda \begin{bmatrix} \psi(x_1) \\ \cdot \\ \cdot \\ \cdot \\ \psi(x_N) \end{bmatrix}. \end{aligned}$$

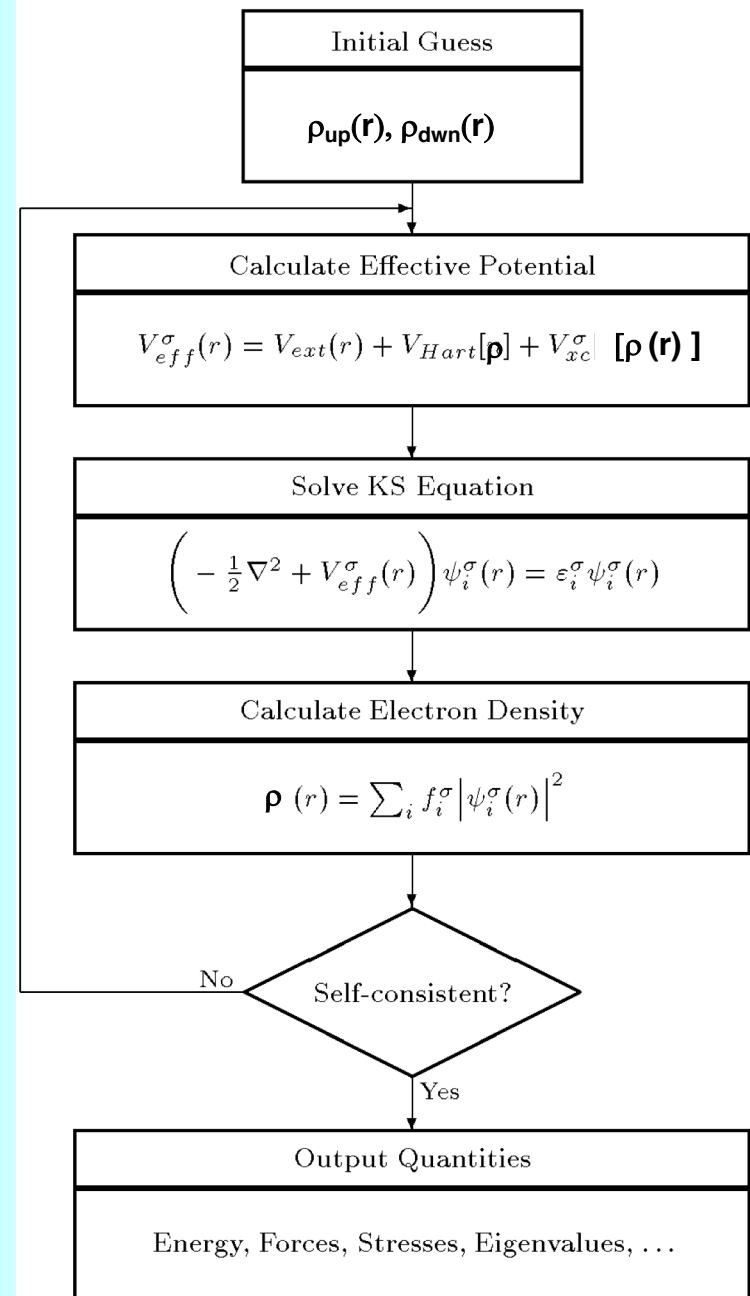
Example: 2-d 4th order Laplacian



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Self-Consistent Kohn-Sham Equations



Iterative solutions of KS eigenvalue problem

- Linear mixing (or following the direction of “steepest descent”)

$$\rho_{i+1}^{\text{in}} = \alpha \rho_i^{\text{out}} + (1-\alpha) \rho_i^{\text{in}} = \rho_i^{\text{in}} + \alpha (\rho_i^{\text{out}} - \rho_i^{\text{in}})$$

- How do we choose α ?

$$\delta \rho = \rho - \rho_{\text{KS}} = \text{deviation from ground state density } \rho_{\text{KS}}$$

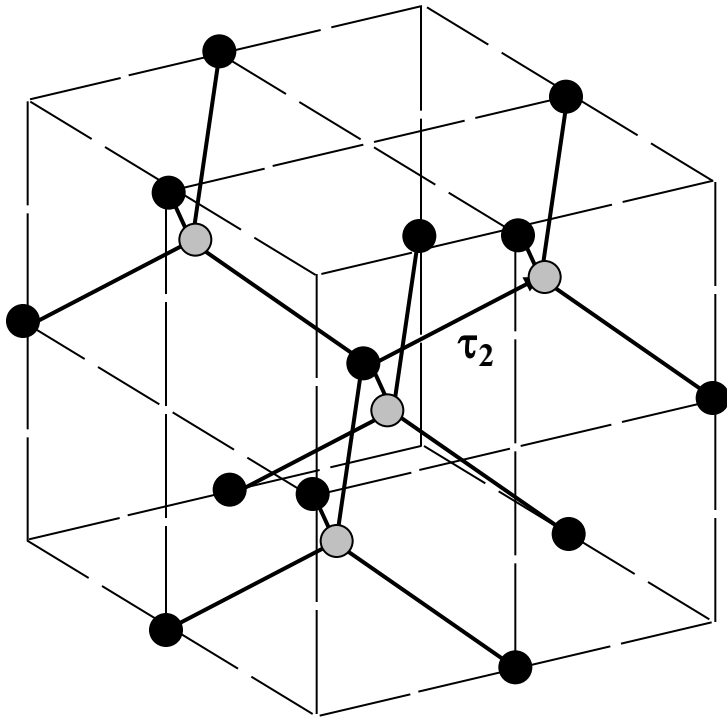
α is related to the response function of a system and to a measure of the polarizability. Linear mixing with large α works better for rigid, strongly bound systems; however convergence can be difficult to obtain for “soft” systems such as a metal surface.

Eigenvalue problem

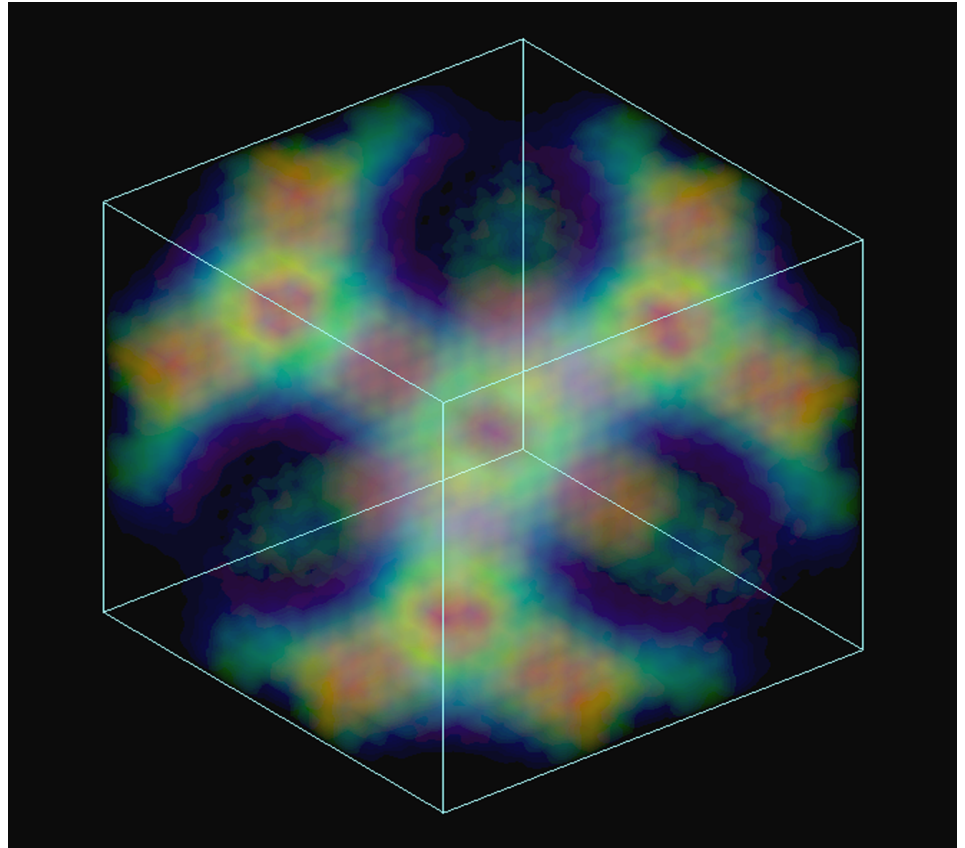
- Direct methods:
 - Full diagonalization of the Hamiltonian matrix in a chosen basis set: **all eigenvalues** $\epsilon_{j,k}$ and **eigenvectors** $\Psi_{j,k}$ are computed
 - In general, suitable for methods with atom-centered, localized orbitals *only*
 - Practical up to a Hamiltonian matrix size of $\sim 10,000$ basis functions
- Variational approaches:
 - Diagonalization problem can be presented as a minimization problem for a quadratic form (the total energy)
 - Steepest descent and conjugate gradients methods

Electron density in solids, e.g. Silicon

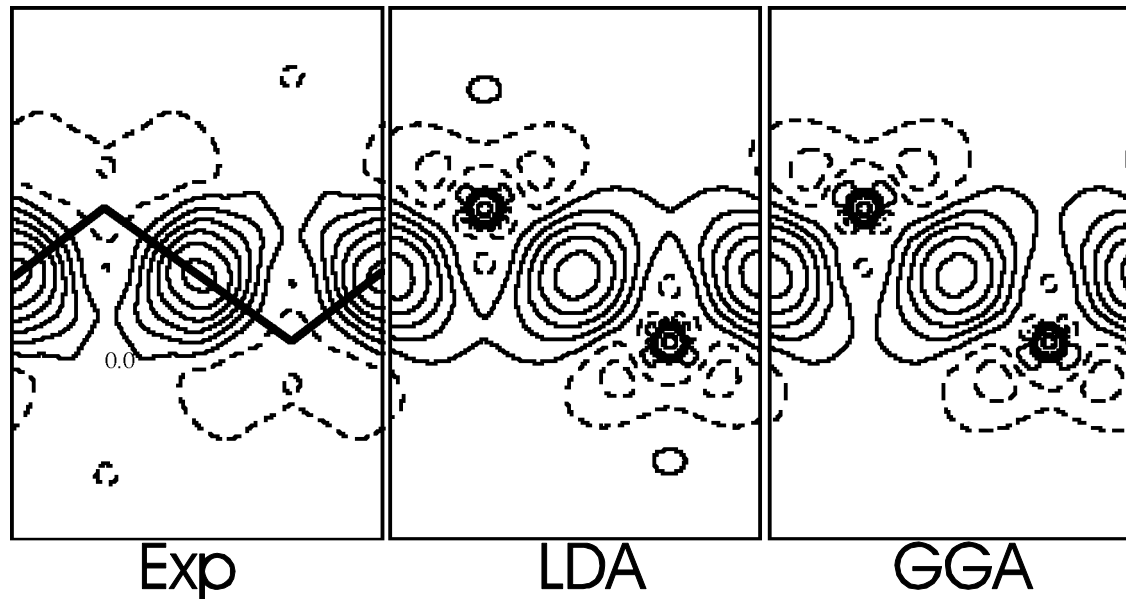
From : "Electronic Structure: Basic Theory and Practical Methods", R. M. Martin, Cambridge University Press, 2004 –



In Si the black and grey atoms are identical



Electron density in Silicon: comparison with experiment



- Electron density difference from sum of atoms
 - Experimental density from electron scattering
 - Calculations with two different functionals and LAPW basis
 - J. M. Zuo, P. Blaha, and K. Schwarz, J. Phys. Cond. Mat. 9, 7541 (1997).
 - Very similar results with pseudopotentials
 - O. H. Nielsen and R. M. Martin (1995)

Comparison of structural parameters obtained using different basis sets.

Method	C		Si		CaF ₂		bcc Fe		
	<i>a</i>	<i>B</i>	<i>a</i>	<i>B</i>	<i>a</i>	<i>B</i>	<i>a</i>	<i>B</i>	<i>m</i>
NCPP ^a	3.54	460	5.39	98	5.21	90	2.75 ^c	226 ^c	
PAW ^a	3.54	460	5.38	98	5.34	100			
PAW ^b	3.54	460	5.40	95	5.34	101	2.75	247	2.00
USPP ^b	3.54	461	5.40	95	5.34	101	2.72	237	2.08
LAPW ^a	3.54	470	5.41	98	5.33	110	2.72 ^d	245 ^d	2.04 ^d
EXP ^a	3.56	443	5.43	99	5.45	85-90	2.87 ^d	172 ^d	2.12 ^d

PW

- *a* – lattice constant; *B* – bulk modulus; *m* – magnetization
- ^aHolzwarth , *et al.*; ^bKresse & Joubert; ^cCho & Scheffler; ^dStizrude, *et al.*