

ESP5403 Nanomaterials for Energy Systems

Size Effect on Transport Phenomena

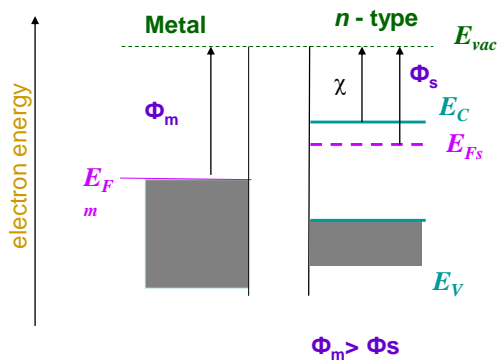
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 6516 7644

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Recall- Schottky barrier

Schottky barrier is formed between a metal and a n - or p - semiconductor when are brought into contact, such that $\Phi_m > \Phi_s$

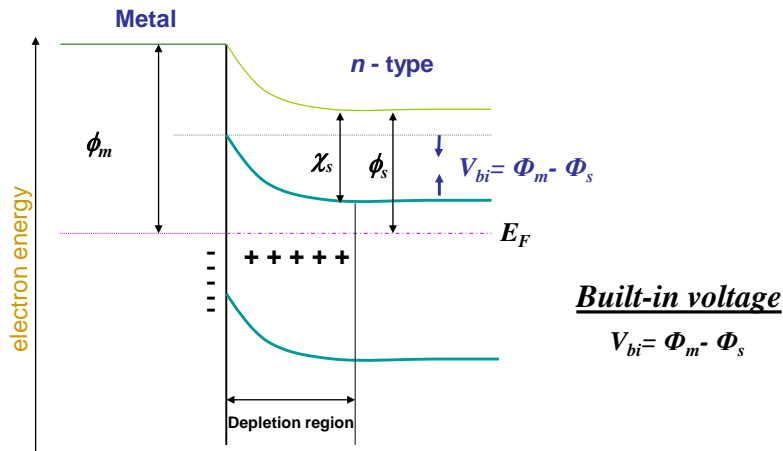


Energy band diagram of the metal and the semiconductor in isolation

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Recall - Schottky barrier (in dark)



Energy band diagram of a metal-semiconductor contact in thermal equilibrium, in the dark

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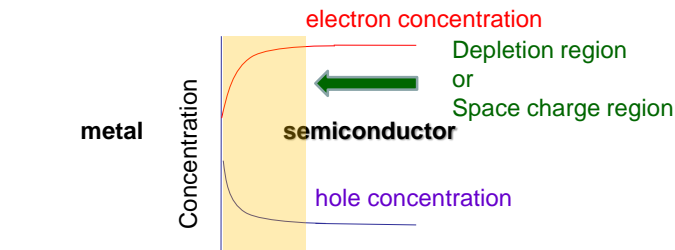
3

Concentration profile at Schottky barrier

Concentration profile

$$n = N_c e^{-(E_c - E_F)/k_B T}$$

$$p = N_v e^{-(E_F - E_v)/k_B T}$$



By joining metal and semiconductor we set up an electric field in a layer close to interface

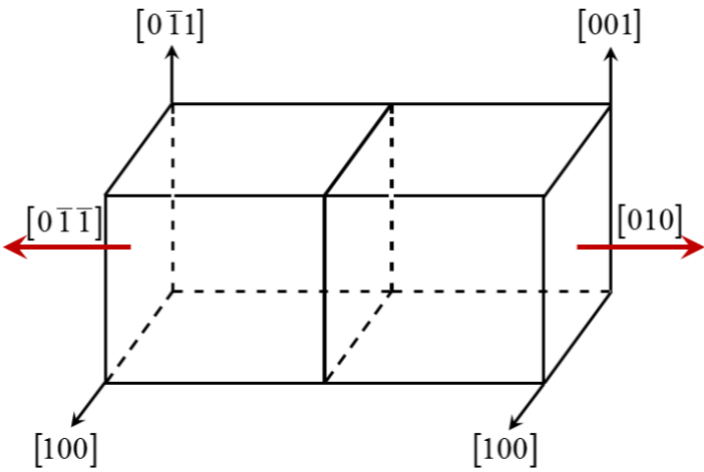
Electric field will drive electrons and holes in opposite direction – separation

Contacts presents a lower resistance path for holes than electrons – from semiconductor to metals – this type of junction is an example for Schottky barrier

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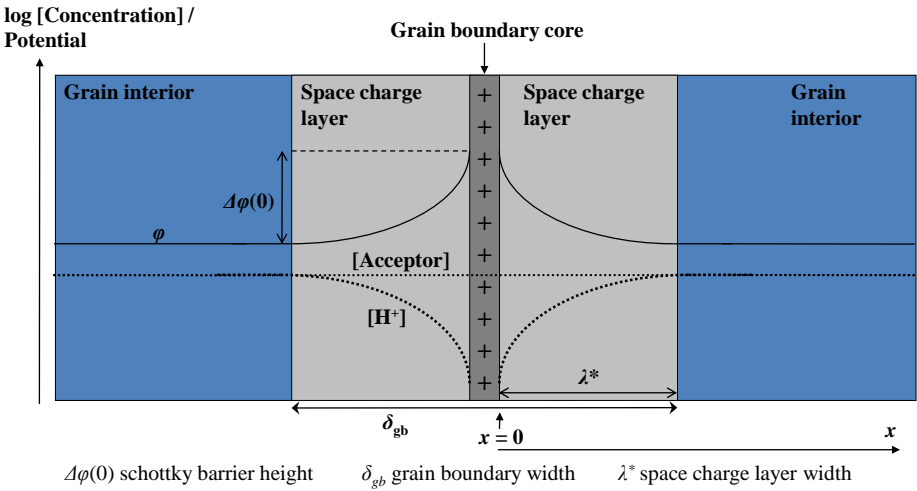
Schematic representation of bi-crystal



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Grain boundary core – space charge model (Mott-Schokky)

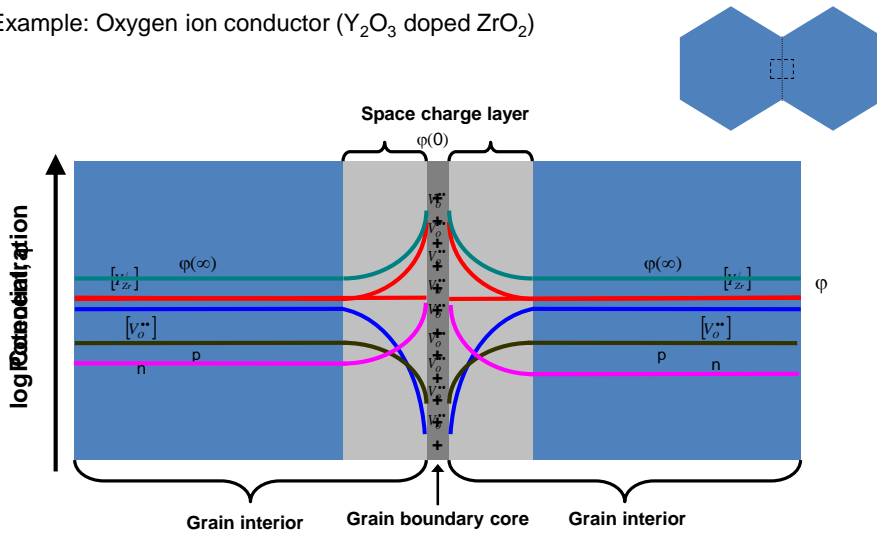


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Grain boundary core – space charge model

Example: Oxygen ion conductor (Y_2O_3 doped ZrO_2)

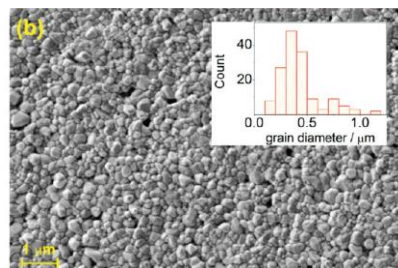
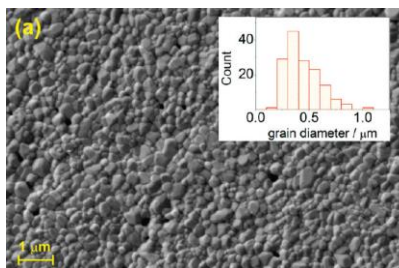
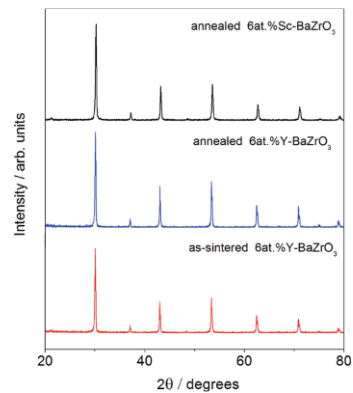


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Dopant segregation and space charge effects in proton-conducting BaZrO_3 perovskites

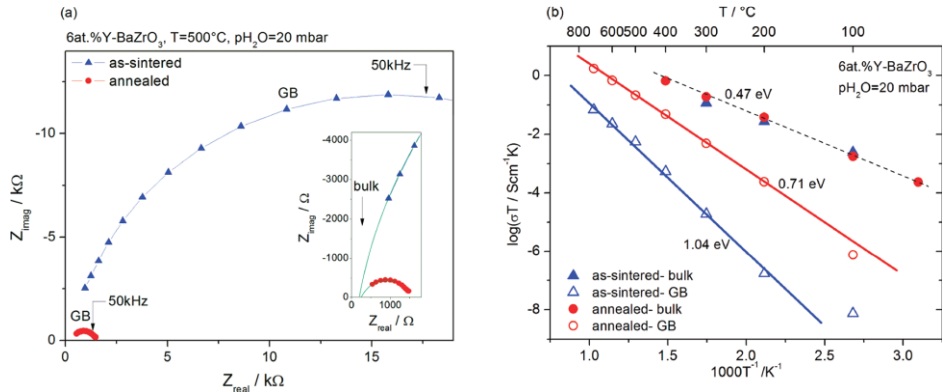
M. Shirpour...J. Maier,
J. Phys. Chem. C 2012, 116, 2453–2461



SEM image and grain size distribution of 6 at %Y-BaZrO₃ (a) as-sintered SPS sample and (b) annealed for 20 h at 1700 C in air.

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Dopant segregation and space charge effects in proton-conducting BaZrO₃ perovskites

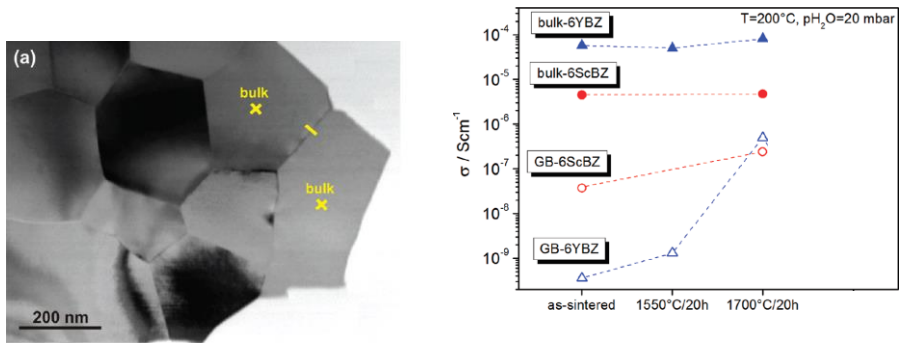


(a) AC impedance spectra at 500 °C and (b) Arrhenius plots of as-sintered and annealed 6 at% Y-doped BaZrO₃ in wet atmosphere (pH₂O= 20 mbar)

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Dopant segregation and space charge effects in proton-conducting BaZrO₃ perovskites

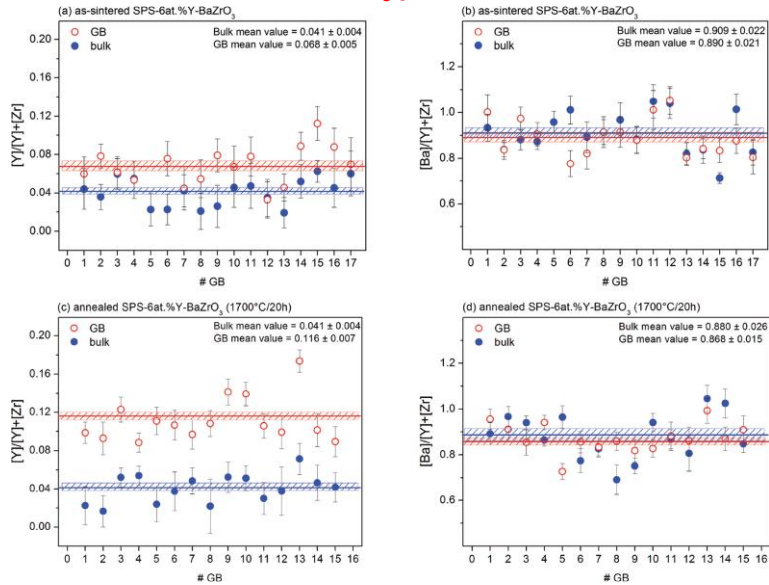


Bulk and grain boundary conductivities of 6 at% Y- and Sc- doped BaZrO₃ under different annealing condition

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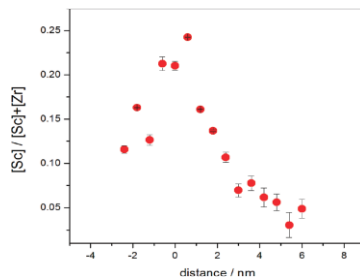
Dopant segregation and space charge effects in proton-conducting BaZrO₃ perovskites



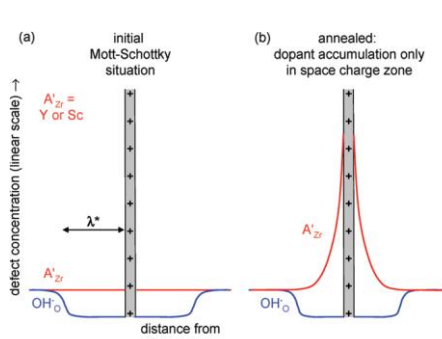
Y and Ba concentration of the GB region compared with the bulk of (a,b) as-sintered and (c,d) annealed 6 at%Y-doped BaZrO₃.

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Dopant segregation and space charge effects in proton-conducting BaZrO₃ perovskites



The local GB composition by transmission electron microscopy with spatially resolved energy dispersive X-ray analysis lines can across a GB of annealed SPS-6 at% Sc-doped BaZrO₃



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Schematic drawing of dopant (A'_{Zr}) and proton concentrations (OH_O) at a grain boundary. (a) As-prepared sample (Mott-Schottky situation: constant dopant concentration, core charge exclusively compensated by proton depletion). (b) After annealing: case of dopant accumulation in the space charge zone (contributing to compensation of core charge) but not in the GB core.

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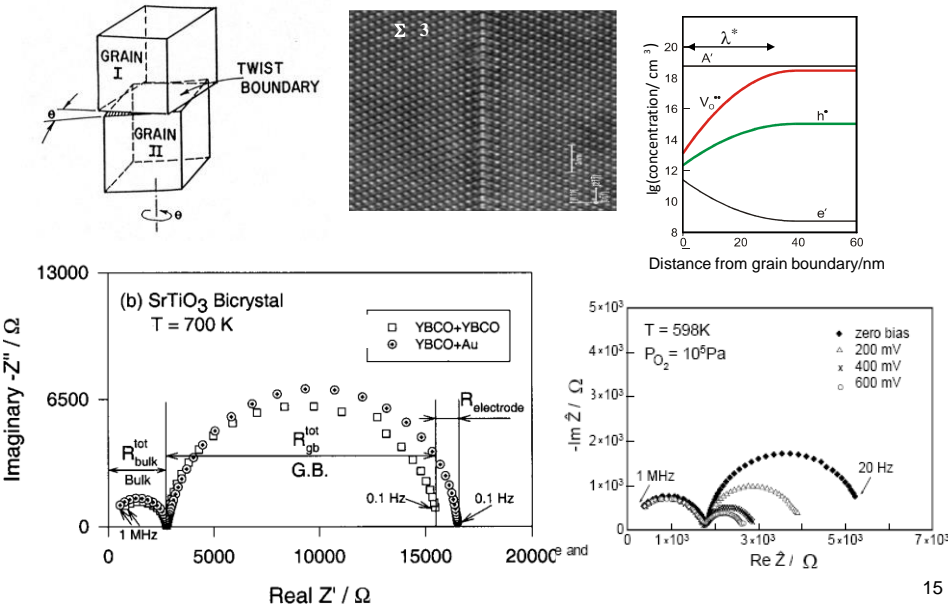
Size Effect on Transport Phenomena

- Depletion of space charges (nanocrystalline SrTiO_2)

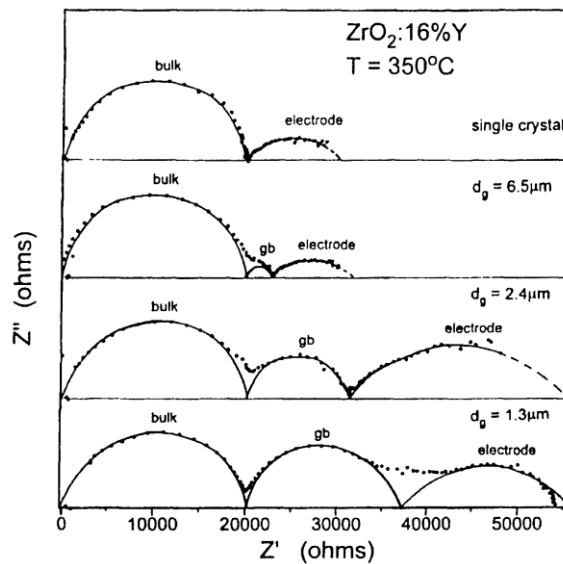
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Electrical conduction in bi-crystal with a single interface – role of interfaces and boundaries

Fe doped SrTiO_3 - bicrystal



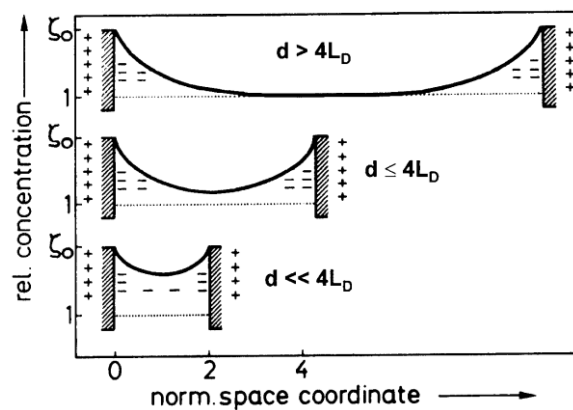
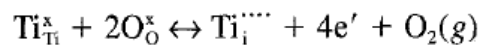
Size effect on grain boundary resistance



Complex impedance spectra obtained for (a) single crystal and microcrystalline YSZ bulk specimens (H.L. Tuller, *Solid State Ionics* 131 (2000) 143-157).

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Size effect on conductivity of TiO_2

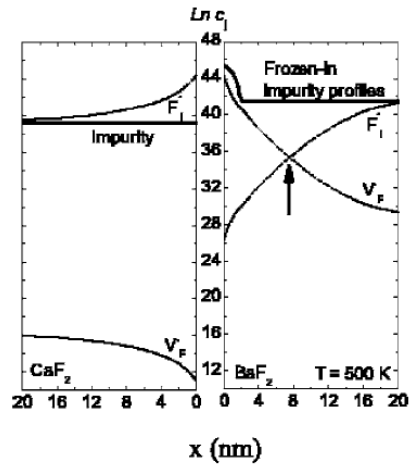


Defect profiles in structures with dimension, d . The build defect concentration is not reached when $d \ll 4L$, where L , is the Debye length

J. Maier, *Solid State Ionics* 23 (1987) 59; C D. Terwilliger and Y.-M. Chiang *J. Am. Ceram. Soc.*, 78, 2045-55 (1995)

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Ionic conduction in $\text{CaF}_2/\text{BaF}_2$ layers

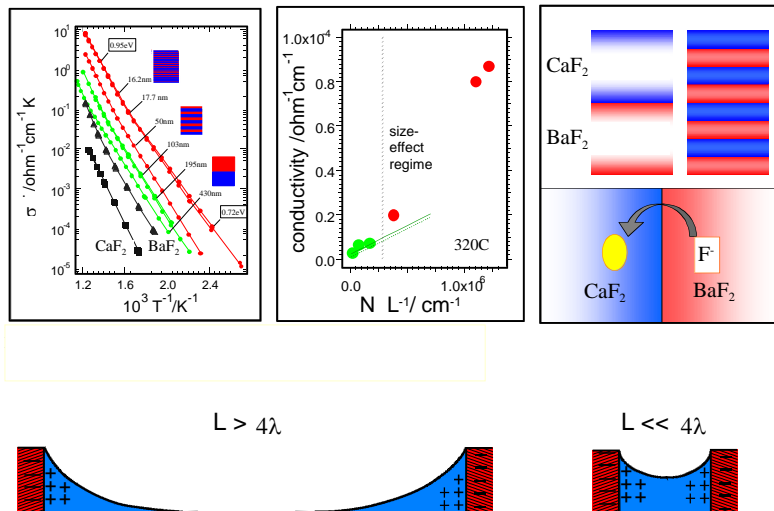


Defect concentration profiles at $T=500\text{ K}$ for $\text{CaF}_2/\text{BaF}_2$ layers. Mott-Schottky profiles with frozen-in impurity (concentration of impurity varies) are shown for BaF_2 , while Gouy-Chapman profiles (concentration of impurity remains constant) for CaF_2 .

Ion conduction across nanosized $\text{CaF}_2/\text{BaF}_2$ multilayer heterostructures;
X.X. Guo...J. Maier, Appl. Phys. Lett. 91, 103102 (2007)

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Artificial mesoscopic ion conductors: accumulation situation

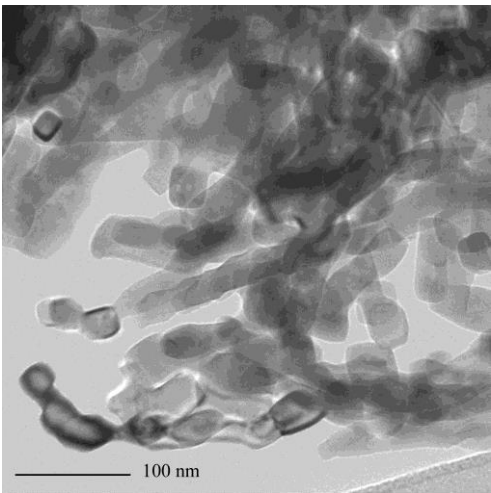


N. Sata, K. Eberman, K. Eberl and J. Maier, *Nature* **408** (2000) 946

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Nanocrystalline SrTiO_3



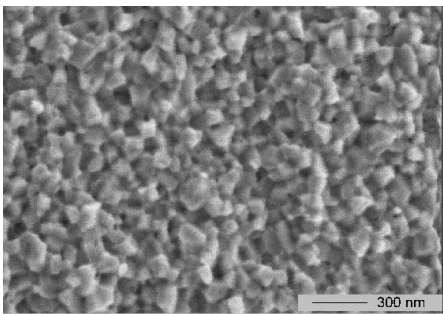
Co-precipitation method
Calcination at 1275 K for 1 hr
Average grain size: $(30 \pm 5 \text{ nm})$
XRD - confirmed single phase formation
 $\text{Sr/Ti} = 1.004$
Total amount of electrochemically active impurities present (ICP):
 $\sim 100 \text{ ppm (iron)}$

TEM image of SrTiO_3 nanopowder

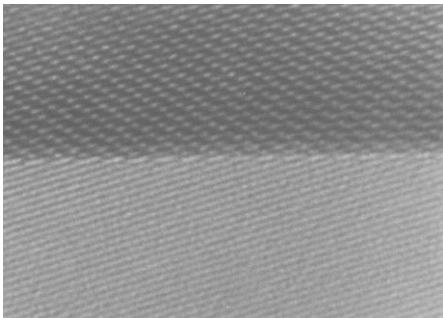
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Nanoceramic SrTiO_3



FESEM image - fractured surface,
Density : 93 %, Average grain size: 80 nm

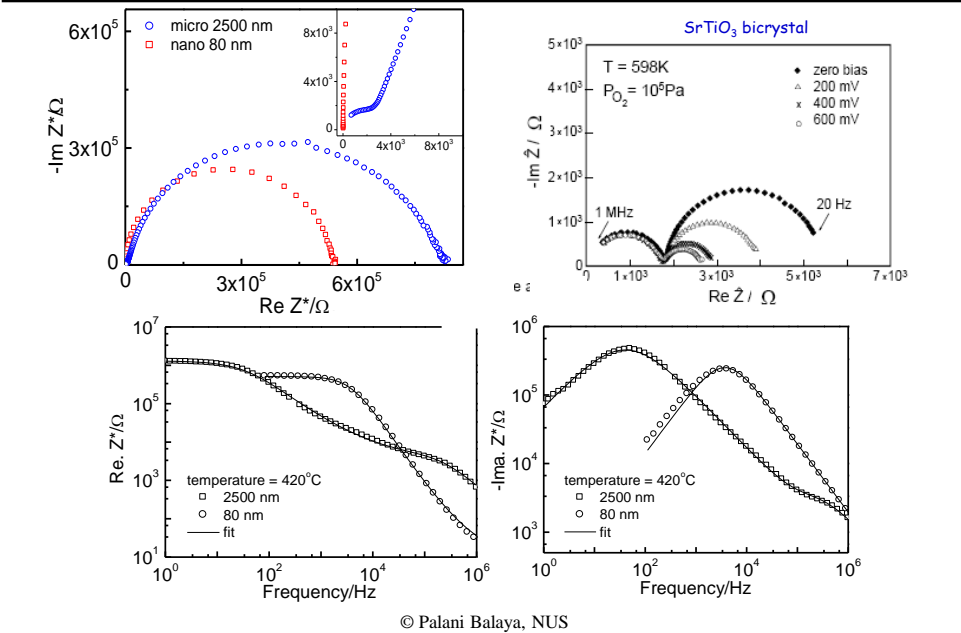


Edge-on view of grain boundary - HRTEM
No amorphous phase - during sintering

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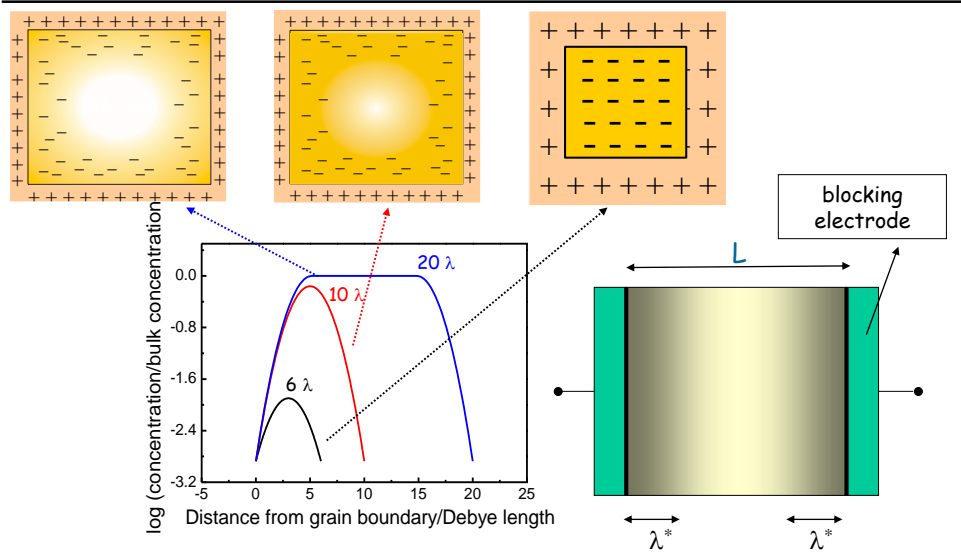
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Impedance spectra of nanocrystalline SrTiO_3



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Mesoscopic (depletion) situation



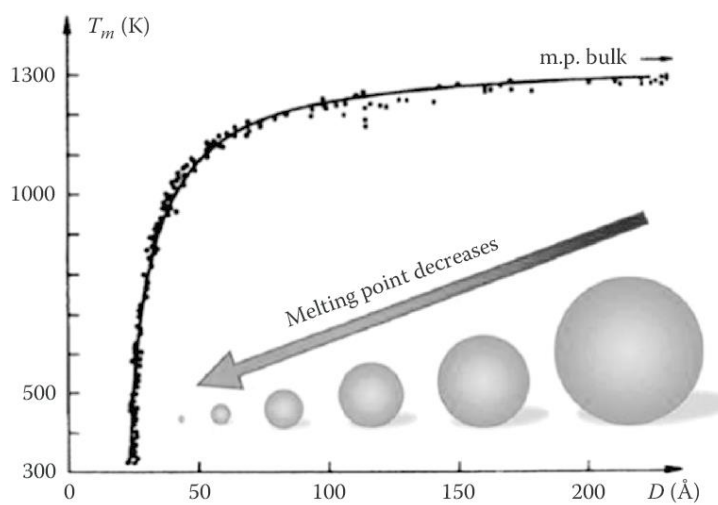
P. Balaya, J. Jamnik, J. Fleig & J. Maier, *Appl. Phys. Lett.*, **86** (2006) 062109; P. Balaya, J. Jamnik, J. Fleig & J. Maier *J. Electrochem. Soc.* **154** (2007) P69

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Thermodynamics at Nanosize

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Relationship between melting temperature and particle diameter of gold nanoparticles



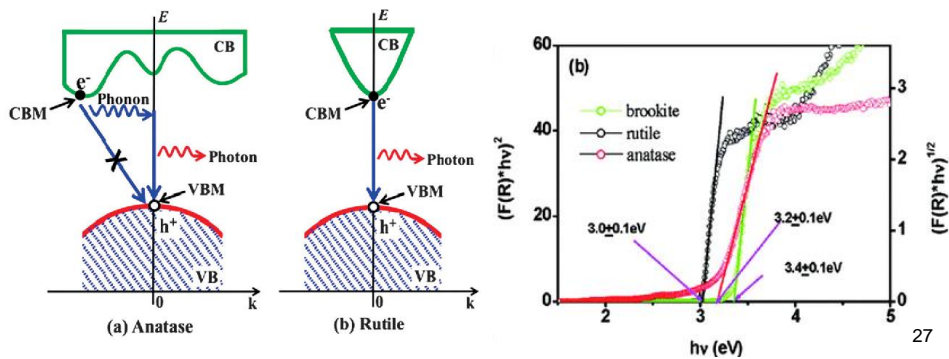
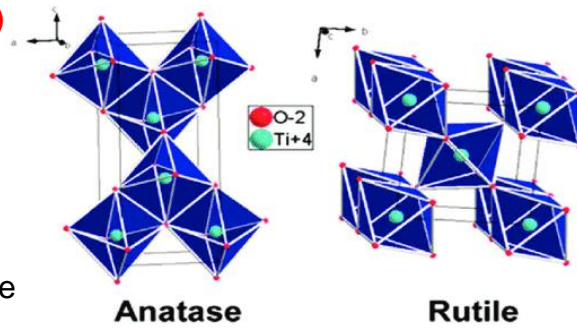
C. Yang et al., J. Mater. Chem. C, 1 (2013) 4052

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Titanium dioxide (TiO₂)

Anatase
(thermodynamically less stable)

Rutile
(thermodynamically more stable)



Energetics of nanocrystalline titania

$$\mu_{MX}^{nano} = \mu_{MX}^{\infty} + 2 \frac{\bar{\gamma}}{\bar{r}} V_m$$

TiO ₂ nano-rutile , Na ₂ Ti ₆ O ₁₃ , Au	β"-Al ₂ O ₃ (Na ₂ O)	TiO ₂ rutile (2 μ), Na ₂ Ti ₆ O ₁₃ , Au
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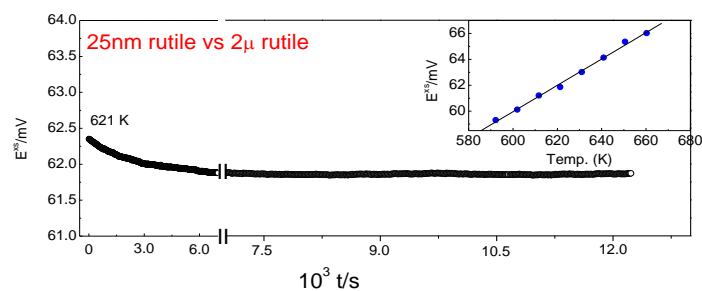
Working electrode

Electrolyte

Reference electrode

Electrode reaction: $\text{Na}_2\text{Ti}_6\text{O}_{13} \rightleftharpoons 2\text{Na}^+ + 2\text{e}^- + 1/2\text{O}_2 + 6\text{TiO}_2$

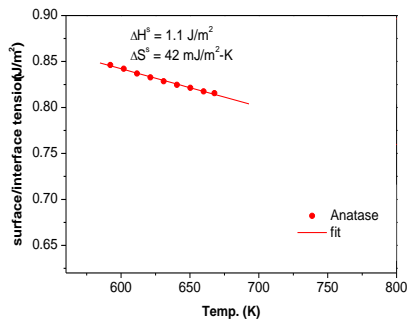
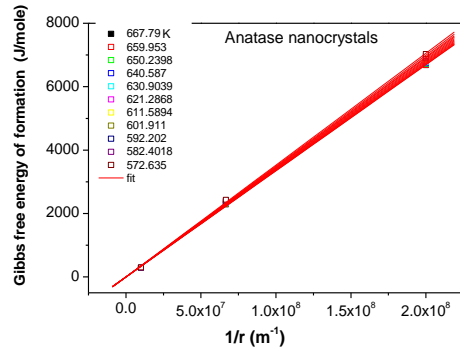
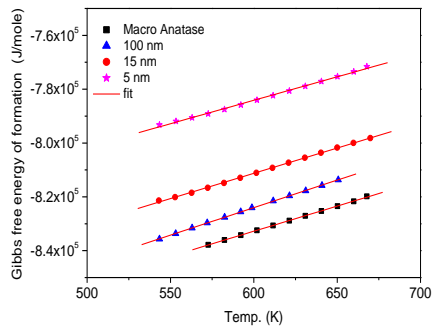
Cell reaction: TiO_2 (nano-rutile) \longrightarrow TiO_2 (micro-rutile)



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Energetics of nanocrystalline titania



$$\Delta G^{xs} = \frac{2\bar{\gamma}}{\bar{r}} V_M = -nFE^{xs}$$

$$\bar{r} \sim 1 \text{ nm}, \bar{\gamma} \sim 1 \text{ Jm}^{-2}$$

$$E^{xs} \sim 100 \text{ mV}$$

P. Balaya and J. Maier, *Phys. Chem. Chem Phys.* 12 (2010) 215.

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