

# ESTIMATING THE AVERAGE SIZE OF FIBER/MATRIX INTERFACE CRACKS IN UD AND CROSS-PLY LAMINATES

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**Summary:** *Initiation and propagation of fiber/matrix interface cracks are analyzed in Representative Volume Elements (RVEs) of UD and cross-ply laminates. By studying the distribution of stresses at the fiber/matrix interface in the undamaged case, an estimate of the initial flaw size is derived. By adopting a 2-parameters energy-based criterion for propagation [1], we then proceed to the estimation of the expected debond size in different microstructural arrangements. Finally, the results are compared with microscopic observations available in the literature [2].*

## 1. INTRODUCTION

Characterization of initiation and growth of fiber/matrix interface cracks (or debonds) in Fiber Reinforced Polymer (FRP) composites has proven elusive, particularly the determination of initial flaw size and maximum debond extension. Different methodological approaches exist in the literature to study debond growth, namely Linear Elastic Fracture Mechanics (LEFM) [3], Finite Fracture Mechanics (FFM) [4], Cohesive Zone Modeling (CSM) [5], as well as several initiation physics-based criteria, such as tensile failure of the interface [3], local cavitation-induced brittle matrix failure [6, 7], local matrix plasticization [5], local nucleation plane [8]. We propose here an attempt to evaluate initiation and propagation of debonds by taking into account the uncertainty in failure criteria and mechanisms evident from the literature. We thus aim to estimate a range of plausible values of initial flaw size and maximum debond size by: first, analyzing the stress distribution at the fiber/matrix interface in the absence of damage in Representative Volume Elements (RVEs) of UD and cross-ply laminates; second, determining the angular size of the regions where  $\sigma \geq 99\% \sigma_{max}$ ,  $\sigma \geq 95\% \sigma_{max}$  and  $\sigma \geq 90\% \sigma_{max}$  with  $\sigma_{max}$  the peak stress value, which provides an estimate of the expected initial debond size; third, estimating  $G_{Ic}$  (Mode I critical Energy Release Rate, one of the two parameters in the adopted propagation criterion [1]), by assuming unstable propagation starting from such initial debond size; fourth, identifying the expected maximum debond size in different RUCs and for different values of the material parameters in the propagation criterion [1]. Finally, the estimated range of maximum debond size is compared with the results of microscopical observations of cross-ply specimens reported in [2].

## 2. RVE MODELS AND FE DISCRETIZATION

In this contribution, we analyze debond initiation and propagation in Representative Volume Elements (RVEs) of Uni-Directional (UD) composites and  $[0_{m \cdot k \cdot 2L}^\circ, 90_{k \cdot 2L}^\circ, 0_{m \cdot k \cdot 2L}^\circ]$  laminates. Given a global reference frame with axis  $x$ ,  $y$  and  $z$ , both types of composites are modeled as plates lying in the  $x - y$  plane, with the through-the-thickness direction thus aligned with the  $z$  axis. The UD composite  $0^\circ$  direction is parallel to the  $y$  axis, while the cross-ply  $0^\circ$  direction is parallel to the  $x$  axis. Both composites are loaded in tension along the  $x$  axis, which thus corresponds to: transverse loading of the UD specimen; axial loading of the cross-ply specimen. In both composites, damage is present only in the form of fiber/matrix interface cracks, or debonds. In cross-ply, debonds are assumed to be present only in the central  $90^\circ$  ply. Given that: first, in the presence of a load in the  $x$ -direction, in both lay-ups the  $y$ -strain due to Poisson's effect is very small; second, debond size is assumed to be considerably larger in the fiber than in the arc direction [9]; third, we are interested in studying debond growth along the arc direction; we can consider  $2D$  models under plane strain conditions defined in the  $x - z$  plane. Although generalized plane strain conditions would represent a more appropriate modeling choice, it would limit the ability to compare with previous results. Simple plane strain conditions are thus preferred. UD composites and  $90^\circ$  plies are characterized by a regular microstructure following a square-packing configuration of fibers, built through the repetition of a one-fiber unit cell along the horizontal and the vertical direction. This unit cell is a square with the center occupied by one fiber of radius  $R_f = 1 \mu m$  and the rest of the element constituted by matrix. The size of the one-fiber unit cell is  $2L \times 2L$ , such that:

$$L = \frac{R_f}{2} \sqrt{\frac{\pi}{V_f}}, \quad (1)$$

where  $V_f$  is the fiber volume fraction, here assumed to be 60%. It is worth to specify at this point that the choice  $R_f = 1 \mu m$  is arbitrary and stems from the fact that ERR, in the contest of a linear elastic solution as the one considered in this article, is proportional to the geometrical dimensions of the model. Simplicity is thus the main reason for this choice. Also,  $V_f$  is always the same in the one-fiber unit cell and the entire RVE, i.e. no fiber clustering is analyzed in this work. In the case of cross-ply laminates, the  $0^\circ$  layer is homogenized with properties evaluated according to the Concentric Cylinders Assembly with Self-Consistent Shear (CCA-SCS) model [10, 11]. A glass fiber-epoxy system is considered for both UD and cross-ply laminates. Material properties are reported in Table 1.

Table 1. Summary of mechanical properties of fiber, matrix and UD layer.

Material	$V_f$ [%]	$E_L$ [GPa]	$E_T$ [GPa]	$G_{LT}$ [GPa]	$\nu_{LT}$ [-]	$\nu_{TT}$ [-]
Glass fiber	-	70.0	70.0	29.2	0.2	0.2
Epoxy	-	3.5	3.5	1.25	0.4	0.4
UD	60.0	43.442	13.714	4.315	0.273	0.465

The use of coupling conditions allows the study of a Repeating Unit Cells (RUC) of reduced size with respect to the corresponding RVE, which translates in a gain in terms of computational time and memory usage during the evaluation of Finite Element (FE) solution. The RVEs studied in this article are reported in Figure 1 with the corresponding RUC highlighted by dashed line (in blue in the online color version) and with symmetry and coupling conditions represented by rollers (♻️). Details about the central one-fiber unit cell are shown in Figure 2. Notice that the analysis, in terms of stresses and Energy Release Rate (ERR), is conducted on this central one-fiber unit cell, both in the case of an undamaged and of a partially debonded fiber.

Nomenclature and main features of the RVEs studied are described in the following.

**$n \times k$  – free, Figure 1(a):** UD composite with thickness  $t_{UD} = k \cdot 2L$ , where  $k$  is the number of fiber “rows” in the vertical (through-the-thickness) direction and  $2L$  the side length of the one-fiber unit cell as defined in Equation 1. Debonds appear only in the central fiber “row” with  $n - 1$  fully bonded fibers in between, where  $n$  is the number of fibers present in the RUC along the horizontal direction.

**$n \times k$  – coupling, Figure 1(b):** UD composite with multiple fiber “rows” containing debonds with  $n - 1$  fully bonded fibers in between, where  $n$  is the number of fibers present in the RUC along the horizontal direction. Fiber “rows” containing debonds are separated along the vertical (through-the-thickness) direction by  $k - 1$  “rows” of fully bonded fibers, where  $k$  is the number of fiber “rows” present in the RUC along the vertical direction. Debonds that are vertically aligned are placed on the same side of their respective fiber. To model this configuration, conditions of coupling of the vertical displacement are applied on the top side.

**$n \times k$  – asymm, Figure 1(c):** same as in  $n \times k$  – coupling, but debonds that are vertically aligned are placed on opposite sides of their respective fiber. To model this configuration, the following set of conditions is applied to the upper boundary:

$$\begin{aligned} u_z(x, kL) - u_z(0, kL) &= -(u_z(-x, kL) - u_z(0, kL)) \\ u_x(x, kL) &= -u_x(-x, kL) \end{aligned} \quad (2)$$

which represent conditions of anti-symmetric coupling [12].

**$n \times k$  –  $m \times t_{90^\circ}$ , Figure 1(d):** Cross-ply laminate with  $90^\circ$  layer thickness  $t_{90^\circ} = k \cdot 2L$  and  $0^\circ$  layer thickness  $t_{0^\circ} = m \cdot t_{90^\circ}$ .  $k$  is the number of fiber “rows” in the vertical (through-the-thickness) direction of the  $90^\circ$  layer and  $2L$  the side length of the one-fiber unit cell as defined in Equation 1. Debonds are present only in the central fiber “row” of the  $90^\circ$  layer with  $n - 1$  fully bonded fibers in between, where  $n$  is the number of fibers present in the RUC along the horizontal direction.

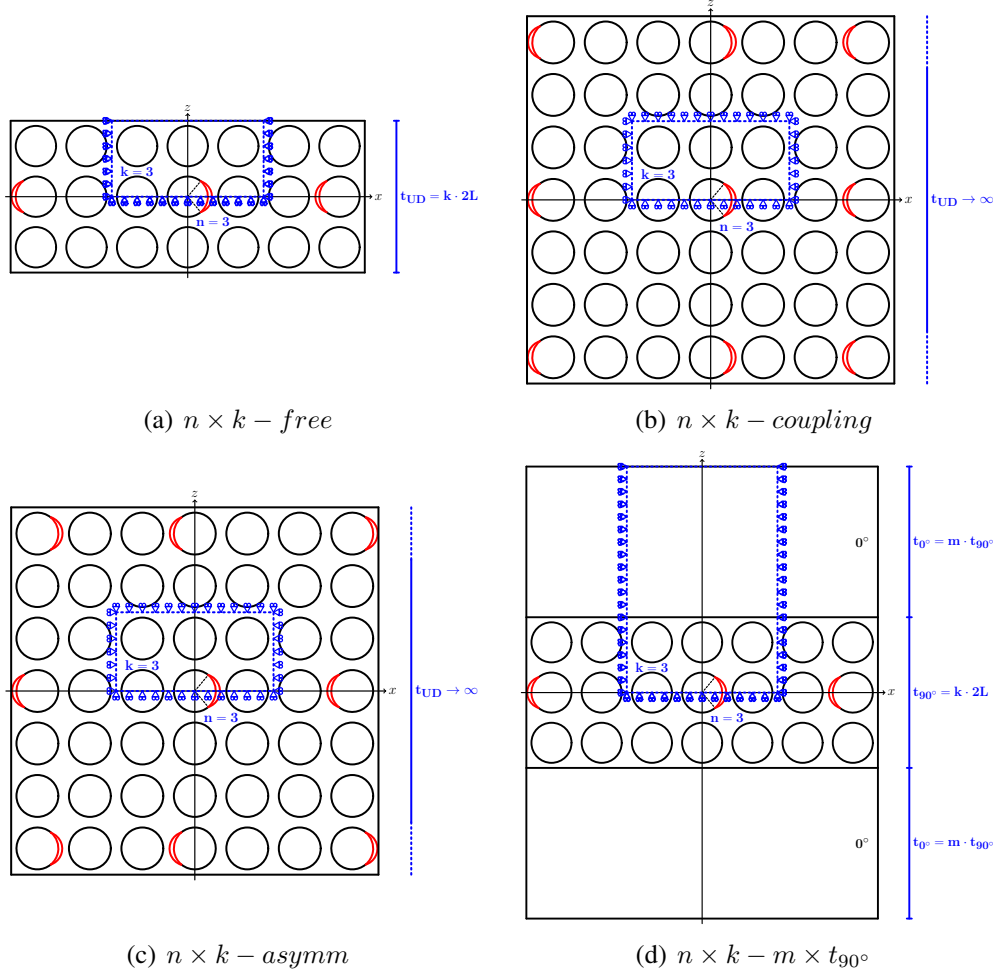


Figure 1. Composite RVEs and corresponding RUCs analyzed.

Every RUC is symmetric with respect to the horizontal ( $x$ ) direction, thus only half of it is modeled in the FE solution through the use of symmetry boundary conditions on the bottom side. Conditions of coupling of the horizontal displacement are applied on the left and right side, to model the repetition of the RUC along the horizontal direction. A tensile load is applied on the right and left side in the form of displacement  $\bar{u}_x = \pm \bar{\varepsilon}_x nL$  with  $\bar{\varepsilon}_x = 1\%$ . The debond has a size of  $2\Delta\theta$  (see Figure 2), with  $\Delta\theta \geq 0$  ( $\Delta\theta = 0$  is the case of no damage at all). For large debonds ( $\Delta\theta \geq 60^\circ - 80^\circ$ ), a region called *contact zone*, of size  $\Delta\Phi$  to be determined by the solution itself, appears at the crack tip. Correct resolution of this behavior requires the imposition of conditions of non-interpenetration of the crack faces. Crack face contact is assumed to be frictionless.

The FE solution is obtained using Abaqus [13] using second order, 2D, plane strain triangular (CPE6) and rectangular (CPE8) elements. To accurately resolve the singularity at the crack tip, a regular mesh of only rectangular elements is used with almost unitary aspect ratio

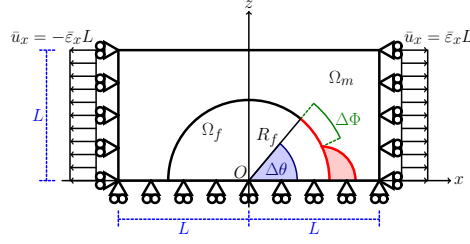


Figure 2. One-fiber unit cell and main parameters characterizing the debonding process.

and angular size  $\delta = 0.05^\circ$ . The crack faces are represented as element-based surfaces with frictionless small-sliding contact pair interaction.

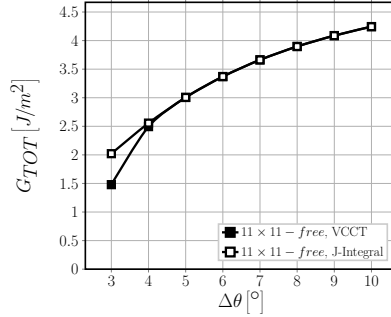


Figure 3. Total ERR of  $11 \times 11 - free$  evaluated respectively with the VCCT and the J-integral.

Total ERR is underestimated, due to the high  $\frac{\delta}{\Delta\theta}$  ratio, with the Virtual Crack Closure Technique (VCCT) [14] for very small debonds (see Figure 3), where it is dominated by Mode I. Thus, for  $\Delta\theta < 10^\circ$ ,  $G_{TOT}$  is evaluated using the J-integral [15],  $G_{II}$  with the VCCT and  $G_I = G_{TOT} - G_{II}$ ; for  $\Delta\theta \geq 10^\circ$  only the VCCT is used. Validation of the model is performed with respect to BEM results of [3, 16]; the order of accuracy of the results is discussed in [17].

### 3. STRESS-BASED ANALYSIS OF DEBOND INITIATION ( $\Delta\theta = 0^\circ$ )

In this section, we assume stress-based debond initiation at the undamaged fiber/matrix interface. Thus, we analyze the distribution of stresses at the fiber/matrix interface in models  $1 \times k - free$  and  $1 \times k - 1 \cdot t_{90^\circ}$  with  $k = 1, 3, 11, 201$ ,  $\Delta\theta = 0^\circ$  (i.e. the undamaged case) and  $\bar{\epsilon}_x = 1\%$ . Some selected stress components are presented, based on their relevance in previous studies on debond initiation and growth: the radial  $\sigma_{rr}$  and shear  $\tau_{r\psi}$  stress [4] (Fig. 4(a) and Fig. 4(b)), the Local Hydrostatic Stress (LHS, following the notation of [18])  $\sigma_{LHS}$  [6, 7] (Fig. 4(c) and Fig. 4(d)), the local von Mises stress  $\sigma_{vM}$  [5] (Fig. 4(e) and Fig. 4(f)), the Local Maximum Principal Stress (LMPS, following the notation of [18])  $\sigma_{LMPS}$  [8] (Fig. 4(g) and Fig. 4(h)). In plane strain conditions imply that there exists an out-of-plane axial component of the stress ( $\sigma_{yy}$  in our notation): in order to study the importance of this out-of-plane component (tri-axial stress state, see [19]),  $\sigma_{LHS}$ ,  $\sigma_{vM}$  and  $\sigma_{LMPS}$  are evaluated both neglecting (index  $2D$ ) and considering it (index  $3D$ ).

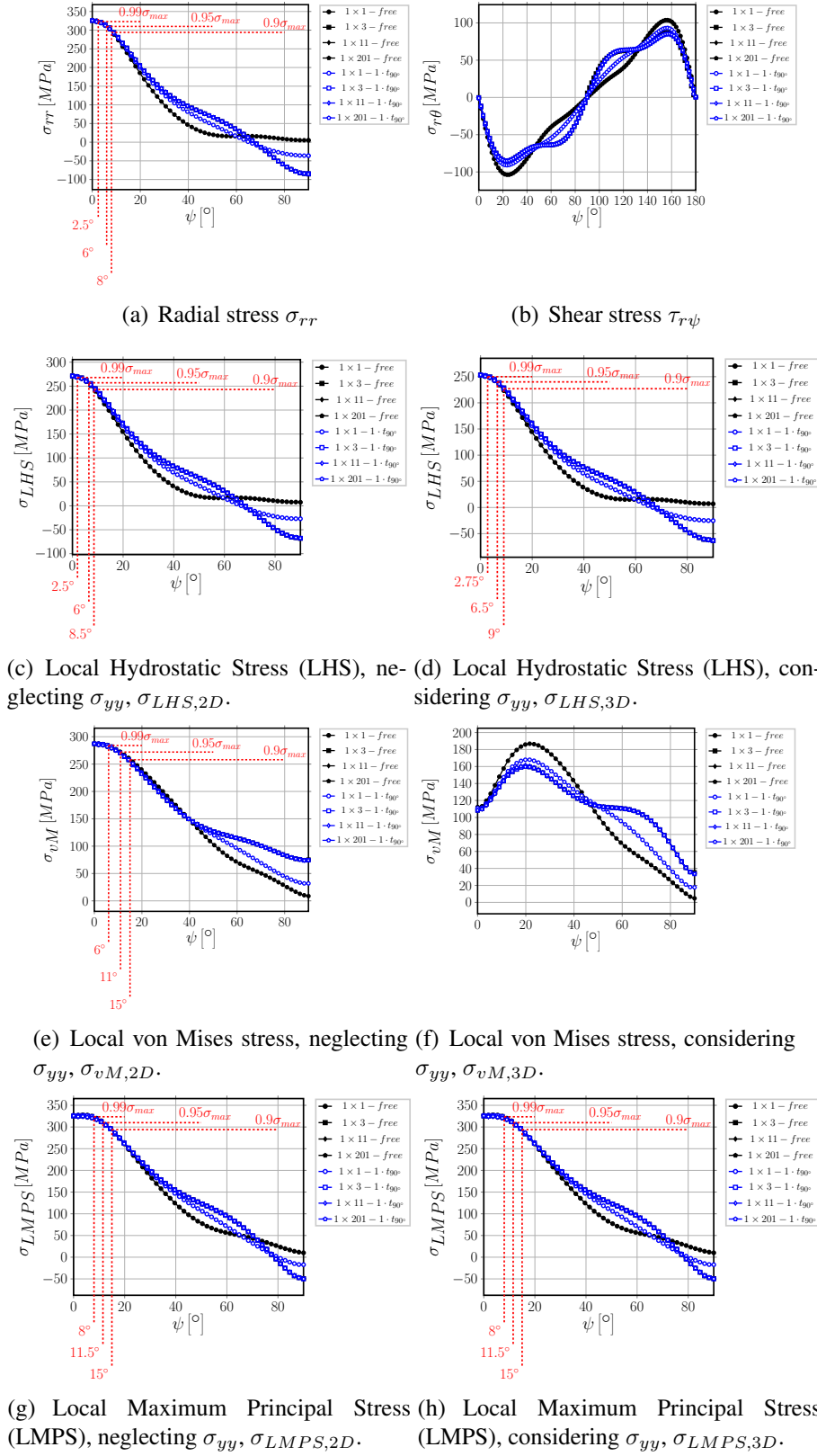


Figure 4. Stress distribution at the fiber/matrix interface in the absence of damage with respect to angular coordinate  $\psi$ ; applied strain  $\bar{\epsilon}_x = 1\%$ .

It is possible to observe that:

1. for all stresses reported in Fig. 4, no significant difference is present between the different RUCs for  $\psi \leq 10^\circ$ ;
2. for all stresses reported in Fig. 4, no difference can be observed by increasing  $k$  when  $k \geq 3$ ;
3. for all stresses reported in Fig. 4, no difference can be observed between  $1 \times k - free$  and  $1 \times k - 1 \cdot t_{90^\circ}$  for  $k \geq 3$ ;
4. at  $\psi = 0^\circ$  (i.e. the intersection between fiber/matrix interface and horizontal axis); the shear stress  $\tau_{r\psi}$  (Fig. 4(b)) is 0 while the radial stress  $\sigma_{rr}$  (Fig. 4(a)) is at its maximum;
5. comparison of Fig. 4(c) with Fig. 4(d) shows that the out-of-plane stress  $\sigma_{yy}$  has only a marginal effect on the local hydrostatic stress  $\sigma_{LHS}$ : the distribution remains the same, while the peak value is reduced from 270 [MPa] in Fig. 4(c) to 250 [MPa] in Fig. 4(d) by considering  $\sigma_{yy}$ ;
6. comparison of Fig. 4(e) with Fig. 4(f) shows a remarkable effect of the out-of-plane stress  $\sigma_{yy}$  on the von Mises stress: if  $\sigma_{yy}$  is neglected,  $\sigma_{vM}$  presents a maximum of 287 [MPa] at  $\psi = 0^\circ$ , slightly higher than the value of  $\sigma_{LHS}$  (neglecting  $\sigma_{yy}$ ) of 270 [MPa] at  $\psi = 0^\circ$ ; if  $\sigma_{yy}$  is considered, the peak value of  $\sigma_{vM}$  is shifted to  $\psi \sim 20^\circ$  and reduced to 186 [MPa] while at  $\psi = 0^\circ$   $\sigma_{vM}$  is reduced to  $\sim 110$  [MPa], significantly lower than  $\sigma_{LHS} = 250$  [MPa] at  $\psi = 0^\circ$  (considering  $\sigma_{yy}$ );
7. comparison of Fig. 4(g) with Fig. 4(g) shows that the Local Maximum Principal Stress is practically unaffected by the out-of-plane stress  $\sigma_{yy}$ ;
8.  $\sigma_{rr}$ ,  $\sigma_{LHS,2D}$ ,  $\sigma_{LHS,3D}$ ,  $\sigma_{vM,2D}$ ,  $\sigma_{LMPS,2D}$  and  $\sigma_{LMPS,3D}$  all reach their peak value at  $0^\circ$  and  $180^\circ$  and decrease to 99% the peak value between  $2^\circ$  and  $8^\circ$ , to 95% the peak value between  $6^\circ$  and  $12^\circ$  and to 90% the peak value between  $8^\circ$  and  $15^\circ$  from the occurrence of the maximum.

Based on the previous considerations, it is reasonable to assume that a stress-based criterion would predict, irrespectively of the specific criterion chosen, the onset of an interface crack at  $0^\circ$  or  $180^\circ$  with an initial size at least comprised in the range  $2^\circ - 8^\circ$  (1% margin) and likely in the range  $6^\circ - 12^\circ$  (5% margin). It is worth to point out that, in the case of a linear elastic solution as the one presented here, the stresses at the fiber/matrix interface for a different value of the applied strain  $\bar{\epsilon}_x$  will change in magnitude but will keep the same functional form and relative magnitude with respect to each other. Thus, the considerations made so far, based on the relative magnitude of stresses and their rate of decrease from the peak value, will apply at any other strain level.

#### 4. ENERGY-BASED ANALYSIS OF DEBOND PROPAGATION

In the previous section, we assumed stress-based debond initiation. We now assume that (as in [20]), once an initial debond of a size in the range predicted in the previous section

( $\Delta\theta_0 \sim 2^\circ - 12^\circ$ ) is formed, the debond will grow unstably (at the same level of global applied strain  $\bar{\varepsilon}_x$ ) according to an energy-based criterion of the form [1, 4]:

$$G_{TOT} \geq G_c = G_{Ic} \left( 1 + \tan^2((1 - \lambda) \Psi_G) \right), \quad \Psi_G = \tan^{-1} \left( \sqrt{\frac{G_{II}}{G_I}} \right), \quad (3)$$

where  $\Psi_G$  is the energy-based phase angle,  $G_{TOT}$  is the total ERR,  $G_c$  is the critical ERR,  $G_{Ic}$  is the Mode I critical ERR (material property) and  $\lambda$  the mode mixity sensitivity parameter (material property, usually  $0.2 \leq \lambda \leq 0.35$ )<sup>1</sup>. Given that debond propagation starts as soon as an initial debond is formed (in our model), the propagation criterion in Eq. 3 is satisfied for the initial debond size  $\Delta\theta_0$ . Thus, by calculating  $G_I$ ,  $G_{II}$  and  $G_{TOT}$  and equating  $G_c$  to  $G_{TOT}$  in Eq. 3, it is possible to estimate  $G_{Ic}$  for different values of the initial debond size  $\Delta\theta_0$  and of the parameter  $\lambda$  (Figure 5). The estimation is performed using a restricted set of RUCs:  $11 \times 1 - free$ ,  $11 \times 11 - free$ ,  $11 \times 1 - 1 \cdot t_{90^\circ}$  and  $11 \times 11 - 1 \cdot t_{90^\circ}$ , with  $\bar{\varepsilon}_x = 1\%$ .

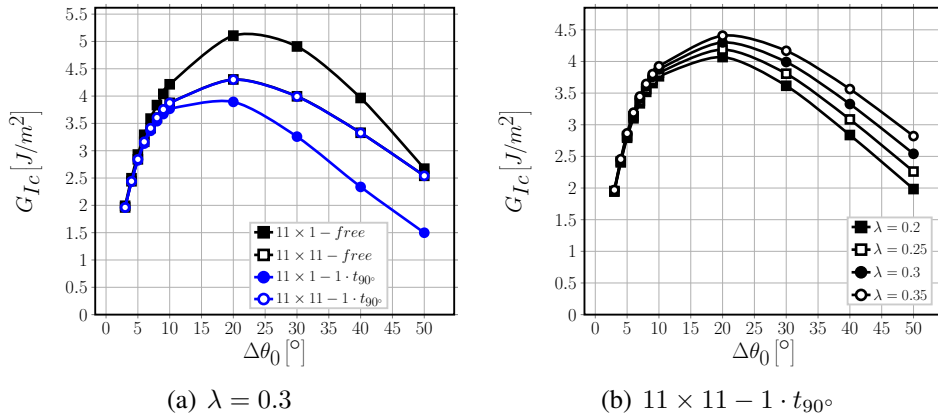


Figure 5. Estimated values of  $G_{Ic}$  assuming different values of initial debond size, for different RUCs and varying  $\lambda$ , with global applied strain  $\bar{\varepsilon}_x$  equal to 1%.

Neither RUC configuration nor  $\lambda$  affects  $G_{Ic}$  significantly for small initial debond sizes, i.e.  $\Delta\theta_0 < 10^\circ$  (Figure 5), corresponding to the range of initial debond size determined in Section 3. Observing Figure 5, we can assume a value of  $G_{Ic}$  in the range  $2 - 4.5 \left[ \frac{J}{m^2} \right]$ , with  $0.2 \leq \lambda \leq 0.35$ . Notice that this estimate is based on calculations with  $\bar{\varepsilon}_x = 1\%$  and the value of  $G_{Ic}$  will scale with  $\bar{\varepsilon}_x$  according to  $\frac{G_{Ic, \bar{\varepsilon}_x, 1}}{G_{Ic, \bar{\varepsilon}_x, 2}} = \left( \frac{\bar{\varepsilon}_{x, 1}}{\bar{\varepsilon}_{x, 2}} \right)^2$ . Given that no dependence of  $G_{Ic}$  on RUC type is observed for  $\Delta\theta_0 < 10^\circ$ , we can use  $G_{Ic} \sim 2 - 4.5 \left[ \frac{J}{m^2} \right]$  to estimate the expected maximum size of debonds in several other RUCs (Figure 6), by evaluating  $G_c$  according to Equation 3. The expected maximum debond size  $\Delta\theta_{max}$  is the value of  $\Delta\theta$  such that

$$G_{TOT}(\Delta\theta \geq \Delta\theta_{max}) \leq G_c(\Delta\theta \geq \Delta\theta_{max}). \quad (4)$$

<sup>1</sup> $\lambda$  governs the influence of mode ratio  $\frac{G_{II}}{G_I}$  on  $G_c$  such that: for  $\lambda = 1$ ,  $G_c = G_{Ic}$ ; for  $\lambda = 0$ ,  $G_c = G_{Ic} \left( 1 + \frac{G_{II}}{G_I} \right)$ .



Notice that, although evaluated using  $G_I$ ,  $G_{II}$  and  $G_{TOT}$  computed with  $\bar{\varepsilon}_x = 1\%$ , the values of  $\Delta\theta_{max}$  do not depend on  $\bar{\varepsilon}_x$  as both sides in Equation 4 scale with the applied strain by a factor  $\left(\frac{\bar{\varepsilon}_{x,1}}{\bar{\varepsilon}_{x,2}}\right)^2$ . If the procedure described in this section is applied using values of  $G_I$ ,  $G_{II}$  and  $G_{TOT}$  computed with a different value of  $\bar{\varepsilon}_x$ , the estimated values of  $\Delta\theta_{max}$  will be the same.

Comparison of Fig. 6(a) with Fig. 6(b) shows that the effect of the presence of the  $0^\circ$  layer is to reduce the maximum size of debonds: from  $\sim 105^\circ$  in  $21 \times 1 - free$  to  $\sim 45^\circ$  in  $21 \times 1 - 1 \cdot t_{90^\circ}$ , for  $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ . However, the presence of just 1 “row” of undamaged fibers between the debond and the free surface or the  $0^\circ/90^\circ$  interface (in RVEs respectively of UD composites and cross-ply laminates) makes the estimate of the maximum debond size unaffected by the number  $k$  of fiber rows (Figures 6(c), 6(d), 6(e) and 6(f)). Moreover, for any given combination of  $G_{Ic}$  and  $\lambda$ , the estimated debond size is the same for  $n \times k - free$  and  $n \times k - 1 \cdot t_{90^\circ}$  for  $k \geq 3$ . The presence of a debond on the neighboring fiber in the vertical direction (RUCs  $21 \times 1 - coupling$  and  $21 \times 1 - asymm$ ) favors instead the growth of larger debonds (Figures 6(g) and 6(h)), with the largest size achieved when debonds are on opposite sides of consecutive fibers:  $\sim 115^\circ$  in  $21 \times 1 - asymm$  and  $\sim 110^\circ$  in  $21 \times 1 - coupling$ , for  $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ . Considering the different combinations of  $G_{Ic}$  and  $\lambda$ , it is possible to identify the range of maximum debond propagation: for a 1-fiber row UD composite ( $21 \times 1 - free$ ), between  $30^\circ$  ( $G_{Ic} = 4.5 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.0.2$ ) and  $105^\circ$  ( $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ ); for a 1-fiber row  $90^\circ$  ply in a cross-ply laminate ( $21 \times 1 - 1 \cdot t_{90^\circ}$ ), between  $30^\circ$  ( $G_{Ic} = 3 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.2$ ) and  $45^\circ$  ( $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ ); for a  $k$ -fiber rows UD composite and a  $k$ -fiber rows  $90^\circ$  ply in a cross-ply laminate with  $k \geq 3$  ( $21 \times k - free$  and  $21 \times k - 1 \cdot t_{90^\circ}$ ), between  $40^\circ$  ( $G_{Ic} = 3 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.2$ ) and  $60^\circ$  ( $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ ); for consecutive debonds along the vertical direction in a UD composite ( $21 \times 1 - coupling$  and  $21 \times 1 - asymm$ ), between  $80^\circ$  ( $G_{Ic} = 4.5 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.2$ ) and  $110^\circ$  ( $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ ) when debonds are on the same sides and between  $60^\circ$  ( $G_{Ic} = 3 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.2$ ) and  $115^\circ$  ( $G_{Ic} = 2 \left[\frac{J}{m^2}\right]$ ,  $\lambda = 0.35$ ) when on opposite sides. These theoretical predictions are in agreement with the microscopical observations performed in [2] on  $[0_3^o, 90_3^o]_S$  carbon fiber (AS4) - epoxy (8552) specimens subjected to tensile loading (corresponding to RUCs  $n \times k - 1 \cdot t_{90^\circ}$  in this article, for which we estimated the range of maximum debond size to be  $40^\circ - 60^\circ$ ). They report debond sizes in the range  $21.4^\circ - 89.2^\circ$  with average value of  $49.3^\circ$  and standard deviation of  $11.7^\circ$ , with 63% of the measurements in the range  $40^\circ - 60^\circ$ .

## 5. CONCLUSIONS

Onset and propagation of fiber/matrix interface cracks has been studied in Representative Volume Elements (RVEs) of UD composites and  $[0_{m \cdot k \cdot 2L}^o, 90_{k \cdot 2L}^o, 0_{m \cdot k \cdot 2L}^o]$  laminates by means of Repeating Unit Cells (RUCs). Analysis of the stress distribution at the fiber/matrix interface in the absence of damage has shown that a stress criterion for initiation would predict, irrespectively of which criterion from those proposed in the literature is chosen, the onset of a debond at  $0^\circ$  or  $180^\circ$  with a semi-aperture  $\Delta\theta_0$  in the range  $2^\circ - 12^\circ$ , corresponding to a margin

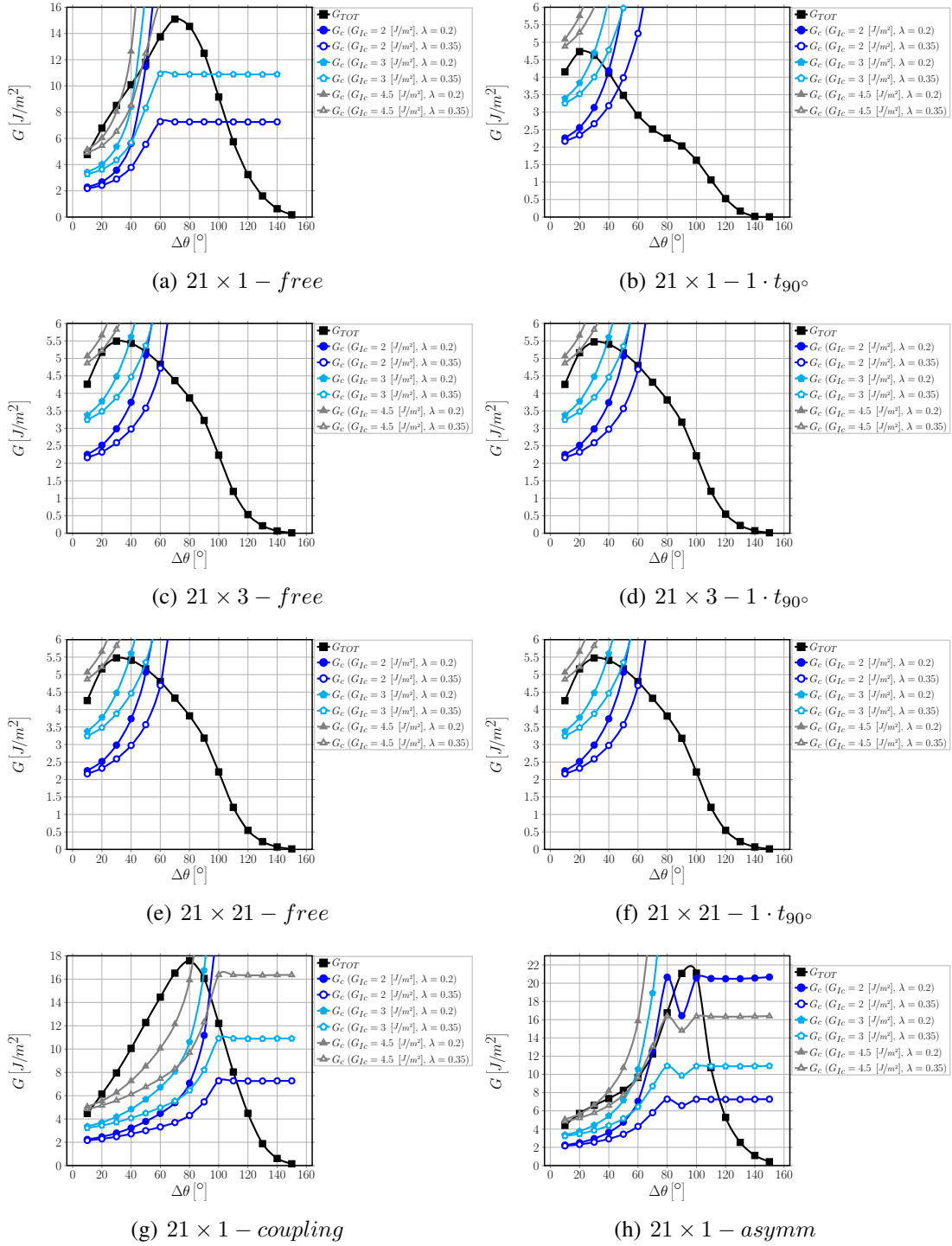


Figure 6.  $G_{TOT}$  and  $G_c$  for several RUCs and different values of  $G_{IC}$  and  $\lambda$ , computed for  $\bar{\epsilon}_x = 1\%$ . Each intersection point of  $G_{TOT}$  and  $G_c$  curves such that  $G_{TOT}(\Delta\theta \geq \Delta\theta_{intersection}) \leq G_c(\Delta\theta \geq \Delta\theta_{intersection})$  determines an estimate of  $\Delta\theta_{max}$ .

of 5% on the satisfaction of the criterion. Assuming that debond propagation occurs unstably immediately after debond onset at the same level of global applied strain  $\bar{\varepsilon}_x$  (as in [20]), it is possible to evaluate the parameter  $G_{Ic}$  in the expression of the critical ERR [1] and with it to estimate the range of expected maximum debond size. The latter finally provides the following ranges: for a 1-fiber row UD composite,  $30^\circ - 105^\circ$ ; for a 1-fiber row  $90^\circ$  ply in a cross-ply laminate,  $30^\circ - 45^\circ$ ; for a multiple fiber-rows UD composite and a multiple fiber-rows  $90^\circ$  ply in a cross-ply laminate,  $40^\circ - 60^\circ$ ; for consecutive debonds along the vertical direction in a UD composite,  $80^\circ - 110^\circ$  when debonds are on the same sides and  $60^\circ - 115^\circ$  when on opposite sides. The prediction for a cross-ply laminate (models  $n \times k - 1 \cdot t_{90^\circ}$ ,  $k \geq 3$ ) agrees well with the debond size distribution estimated in [2] through microscopic observations of  $[0_3^o, 90_3^o]_S$  specimens loaded in tension.

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