

## Molecular Dynamics Simulations

- \* Purely deterministic method.

- \* Follow time evolution of systems by solving Newton's equation of motion:

$$m \frac{d^2 \mathbf{r}}{dt^2} = -\mathbf{F}(\mathbf{r}) \quad \text{where } \mathbf{F}(\mathbf{r}) = -\nabla V(\mathbf{r})$$

- \* Properties of statistical ensemble are calculated as time averages.

- \* Microcanonical ensemble (N, V, E) is the common choice + other constraints:

eg. momentum M =====> (N, V, E, M) = MD ensemble.

- \* For N particles, problem reduces to solving systems of 6N 1<sup>st</sup> order ODE:

$$\begin{cases} \dot{\mathbf{r}}_i &= \mathbf{\bar{v}}_i \\ m\dot{\mathbf{v}}_i &= \mathbf{\bar{F}}_i \end{cases}; \quad \mathbf{F}_i = -\nabla_{\mathbf{r}_{ij}} \sum_{i \neq j} V(\mathbf{r}_{ij}); \quad \nabla_{\mathbf{r}_{ij}} = \hat{\mathbf{i}} \frac{\partial}{\partial x_{ij}} + \hat{\mathbf{j}} \frac{\partial}{\partial y_{ij}} + \hat{\mathbf{k}} \frac{\partial}{\partial z_{ij}}$$

# Potential Functionals

\* **2-body interactions:**  $\Phi(1, \dots, N) = \sum_{i < j} V_2(i, j)$

- **Lennard-Jones (LJ):**

$$V_2(r) = 4\varepsilon [x^{-12} - x^{-6}]$$

$\varepsilon$  - maximal well-depth of potential.

$x = r/\sigma$  - distance in units of  $\sigma$ , where  $\sigma$  is effective atomic radius.

works fine for rare gases, still used for gas – metal interactions  
in thin film growth simulations

- **Yukawa potentials:**

Simple:  $V_2(r) = (E/x) \exp[-a(x-1)]$

Double:  $V_2(r) = (E/x) \{ \exp[-a(x-1)] - \exp[-b(x-1)] \}$

$x$  just as for Lennard-Jones,  $a$  and  $b$  are parameters.

advantages over soft-spheres and LJ to perform “analytical” investigations in thermodynamics and variational free-energy studies.

- Morse:

$$V_2(r) = E \{ \exp[-2a(r - r_0)] - 2\exp[-a(r - r_0)] \}$$

similar properties to LJ.

allows for more intermediate interactions.

behaves more like a bonding-type potential.

- Hard-Sphere:

$$V(r) = \infty; r < \sigma \quad \text{and} \quad V(r) = 0; r > \sigma$$

used initially in first MD simulations

“wall-like” potential, still useful in some contexts.

- Soft-Sphere:

$$V(r) = x - n, \text{ with } x \text{ as for LJ and } n \text{ can vary}$$

$n = 1$ , yields soft interaction.

the larger  $n$ , the harder the repulsion.

- Other potentials: Square-Well, Ionic

## Potential Functionals

\* **3-body interactions:**  $\Phi(1, \dots, N) = \sum_{i < j} V_2(i, j) + \sum_{i < j < k} V_3(i, j, k)$

- **Stillinger – Weber:**

$V_2(r) = A (Br^p - r^q) \exp[(r-a) - 1]$ ; where A, B, p, q, a – params.

$V_3(r) = h(r_{ij}, \mathbf{r}_{ik}, \theta_{jik}) + h(r_{ji}, \mathbf{r}_{jk}, \theta_{ijk}) + h(r_{ki}, \mathbf{r}_{kj}, \theta_{ikj})$ .

$$E = \sum_i \sum_{j>i} \Phi_2(r_{ij}) + \sum_i \sum_{j \neq ik > j} \sum_k \Phi_3(r_{ij}, r_{ik}, \theta_{ijk})$$

$$\Phi_2(r_{ij}) = A_{ij} \varepsilon_{ij} \left[ B_{ij} \left( \frac{\sigma_{ij}}{r_{ij}} \right)^{p_{ij}} - \left( \frac{\sigma_{ij}}{r_{ij}} \right)^{q_{ij}} \right] \exp \left( \frac{\sigma_{ij}}{r_{ij} - a_{ij} \sigma_{ij}} \right)$$

$$\Phi_3(r_{ij}, r_{ik}, \theta_{ijk}) = \lambda_{ijk} \varepsilon_{ijk} [\cos \theta_{ijk} - \cos \theta_{0ijk}]^2 \exp \left( \frac{\gamma_{ij} \sigma_{ij}}{r_{ij} - a_{ij} \sigma_{ij}} \right) \exp \left( \frac{\gamma_{ik} \sigma_{ik}}{r_{ik} - a_{ik} \sigma_{ik}} \right)$$

- **Tersoff:**

$$V_2(r) = f_C(r) + [f_R(r) + \mathbf{b} f_A(r)]; \mathbf{b} = \mathbf{b}(\theta_{ijk})$$

## Tersoff functional

- the 2 & 3 body interactions require extensive CPU time.

$$E = \sum_i^N \Phi_i; \quad \Phi_i = \frac{1}{2} \sum_{j(\neq i)}^N f_c(r_{ij}) [V_R(r_{ij}) - b_{ij} V_A(r_{ij})]$$

where

$$f_c(r_{ij}) = \begin{cases} 1 & \text{if } r_{ij} < R - D \\ \frac{1}{2} - \frac{1}{2} \sin \left[ \frac{\pi}{2} (r_{ij} - D) / D \right] & \text{if } R - D < r_{ij} < R + D \\ 0 & \text{if } r_{ij} > R + D \end{cases}$$

3-body part given by:

$$V_R(r_{ij}) = A \exp(-\lambda_1 r_{ij}); \quad V_A(r_{ij}) = B \exp(-\lambda_2 r_{ij}); \quad b_{ij} = (1 + \beta^n \xi_{ij}^n)^{-1/2n}$$

$$\xi_{ij} = \sum_{k(\neq i, j)}^N f_c(r_{ik}) g(\theta_{ijk}) \exp[\lambda_3^3 (r_{ij} - r_{ik})^3]$$

$$g(\theta_{ijk}) = 1 + c^2 / d^2 - c^2 / [d^2 + (h - \cos \theta_{ijk})^2]$$

A, B,  $\lambda_1$ ,  $\lambda_2$ ,  $\lambda_3$ ,  $\beta$ , n, c, d, h – given parameters; R, D [Å] – interaction range

## Embedded-Atom Method (EAM) – Johnson's functional

- very useful for transition metals interactions (this is the fcc analytical form);
  - there are also separate forms for bcc and hcp structures.

$$E_t = \sum_i F(\rho_i) + \frac{1}{2} \sum_{i,j} \Phi(r_{ij}); \quad \rho_i = \sum_j f(r_{ij})$$

- where:
- $E_t$  = total internal energy.
  - $\rho_i$  = the electron density at atom  $i$  due to all other atoms.
  - $f(r_{ij})$  = the electron density at atom  $i$  due to atom  $j$  as a fct. of distance.
  - $r_{ij}$  = separation distance between atoms  $i$  and  $j$ .
  - $F(\rho_i)$  = the energy to embed atom  $i$  in an electron density  $\rho_i$ .
  - $\Phi(r_{ij})$  = two-body potential between atoms  $i$  and  $j$ .

- can also be used in the following form for all alloy potentials:

$$\Phi^{ab}(r) = \frac{1}{2} \left( \frac{f^b(r)}{f^a(r)} \Phi^{aa}(r) + \frac{f^a(r)}{f^b(r)} \Phi^{bb}(r) \right)$$

- electron density and two-body potential taken as decreasing exponentials:

$$f(r) = f_e \exp[-\beta(r/r_e - 1)]; \quad \Phi(r) = \Phi_e \exp[-\gamma(r/r_e - 1)].$$

-the embedding function is determined by fitting to a universal eq. of state in the form of a Rydberg fct.:

$$F(\rho) = -E_c(1 - \ln x)x - 6\Phi_e y$$

where:  $x = (\rho/\rho_e)^{\alpha/\beta}$ ;  $y = (\rho/\rho_e)^{\gamma/\beta}$ ;  $\alpha = 3(\Omega B/E_c)^{1/2}$ ;  $\gamma = 15(\Omega G/\beta E_{UF})$

Inputs						Model parameters				
Atom	$\Omega$	$E_c$	$E_{UF}$	$\Omega B$	$\Omega G$	$f_e$	$\Phi_e$	$\alpha$	$\beta$	$\gamma$
Cu	11.81	3.54	1.30	10.17	4.05	0.30	0.59	5.09	5.85	8.00
Ag	17.10	2.85	1.10	11.10	3.61	0.17	0.48	5.92	5.96	8.26
Au	16.98	3.93	0.90	17.70	3.29	0.23	0.65	6.37	6.67	8.20
Ni	10.90	4.45	1.70	12.28	6.45	0.41	0.74	4.98	6.41	8.86
Pd	14.72	3.91	1.54	17.92	4.99	0.27	0.65	6.42	5.91	8.23
Pt	15.06	5.77	1.60	26.60	6.12	0.38	0.95	6.44	6.69	8.57

## Modified Embedded Atom Method (MEAM, 2000-2010)

Main idea is to incorporate most fcc, hcp, bcc transition metals, as well as some gases (N, O) under the same analytical “umbrella”.

$$E_{\text{total}} = \sum_i \left[ \sum_{j < i} \Phi_{ij}(r_{ij}) + F_i(\rho_i) \right] \quad \text{where} \quad F_i(\rho_i) = A E_c \frac{\rho_i}{\bar{\rho}_i} \ln \frac{\rho_i}{\bar{\rho}_i}$$

$F_i$  is the embedding function,  $\rho_i$  is the background  $e^-$  density at lattice site  $i$ , and  $\bar{\rho}_i$  is the background  $e^-$  density for a reference structure.

The improvement/novelty, as well as the difficulty, comes from considering the directionality in bonding through a combination of partial electron density terms,  $\rho_i^{(h)}$ ,  $h = 1-3$ , for different angular contributions with weight factors  $t^{(h)}$ .

$$\left( \rho_i^{(0)} \right)^2 = \left[ \sum_{j(\neq i)} S_{ij} \rho_j^{a(0)}(r_{ij}) \right]^2 \quad \left( \rho_i^{(1)} \right)^2 = \sum_{\alpha} \left[ \sum_{j(\neq i)} \frac{r_{ij}^{\alpha}}{r_{ij}} S_{ij} \rho_j^{a(1)}(r_{ij}) \right]^2$$



$$\left(\rho_i^{(2)}\right)^2 = \sum_{\alpha,\beta} \left[ \sum_{j(\neq i)} \frac{r_{ij}^\alpha}{r_{ij}} \frac{r_{ij}^\beta}{r_{ij}} S_{ij} \rho_j^{a(2)}(r_{ij}) \right]^2 - \frac{1}{3} \left[ \sum_{j(\neq i)} S_{ij} \rho_j^{a(2)}(r_{ij}) \right]^2$$

$$\left(\rho_i^{(3)}\right)^2 = \sum_{\alpha,\beta,\gamma} \left[ \sum_{j(\neq i)} \frac{r_{ij}^\alpha}{r_{ij}} \frac{r_{ij}^\beta}{r_{ij}} \frac{r_{ij}^\gamma}{r_{ij}} S_{ij} \rho_j^{a(3)}(r_{ij}) \right]^2 - \frac{3}{5} \sum_{\alpha} \left[ \sum_{j(\neq i)} \frac{r_{ij}^\alpha}{r_{ij}} S_{ij} \rho_j^{a(2)}(r_{ij}) \right]^2$$

$\rho_j^{a(h)}(r_{ij})$  is the atomic  $e^-$  density from atom  $j$  to site  $i$ ,  $r_{ij}^\alpha$  is the component of the distance vector between atoms  $j$  and  $i$ , with  $\alpha = 1, 2, 3$  representing  $x, y, z$ , and  $S_{ij}$  is the screening function between atoms  $j$  and  $i$ .

$$\rho_j^{a(h)}(r) = f_{\text{cutoff}}(r) \exp \left[ -\beta_j^h \left( \frac{r}{r_{j,e}} - 1 \right) \right] \quad \text{with } \beta_j^{(h)} \text{ pot. parameters and } r_{j,e} \text{ nearest equilibrium distance in reference structure.}$$

$$f_{\text{cutoff}}(r) = \begin{cases} 1, & r \leq r_c - \Delta r \\ \left[ 1 - \left( 1 - \frac{r_c - r}{\Delta r} \right)^4 \right]^2, & r_c - \Delta r < r < r_c \\ 0, & r_c \leq r \end{cases}$$

Here  $r_c$  is the cutoff distance and  $\Delta r$  gives the cutoff region

Electron density is defined as:

$$\rho_j = \rho_j^{(0)} G(\Gamma); \quad \text{with} \quad G(\Gamma) = \frac{2}{1 + \exp(-\Gamma)}; \quad \text{and} \quad \Gamma = \sum_h t_j^{(h)} \left[ \frac{\rho_j^{(h)}}{\rho_j^{(0)}} \right]^2$$

where  $t_j^{(h)}$  are adjustable parameters.

The screening function  $S_{ij}$  is defined as the product of the product of the screening factor,  $S_{ijk}(x_{ijk})$ , due to all neighboring  $k$  atoms.

All summations are carried out over a finite range, typically including only 1<sup>st</sup>, or 1<sup>st</sup> and 2<sup>nd</sup> neighbors only.

Method is still evolving. Works reasonably well for **most transition metals** and a selection of gases. However, to push is to extend its transferability to **binary** and **ternary** compounds.

Difficulties arise from the fact that these materials have complex bonding, such as combinations of **ionic-covalent-metallic-type** bonds.

Latest important additions to MEAM family: **TiN, TiC**.

New Parameterization  
of the  
**Modified Embedded Atom Potential**  
for  
Large Scale Simulations  
of **TiN Thin Films Growth**

D.G. Sangiovanni, V. Chirita & L. Hultman

Thin Film Physics, Linköping University, Sweden

AVS 2010, Albuquerque, USA

# TiN MEAM parameterization\*

Fitted to bulk properties

- Lattice parameter
- Cohesive energy
- Elastic constants & moduli
- Thermal linear expansion

Surface properties & related phenomena to be verified

- Surface energies
- Surface relaxation

Critical tests for thin film growth simulations

- Adatom diffusion & Ehrlich barriers

\* YM Kim & BJ Lee, Acta Materialia **56**, 3481 (2008)

# Original parameters

	Single element system													
	$E_c$	$r_e$	$B$	$A$	$\beta^{(0)}$	$\beta^{(1)}$	$\beta^{(2)}$	$\beta^{(3)}$	$t^{(1)}$	$t^{(2)}$	$t^{(3)}$	$C_{min}$	$C_{max}$	$d$
<b>Ti</b>	4.87	2.92	110	0.66	2.70	1.0	3.0	1.0	6.80	-2.0	-12.0	1.00	1.44	0.00
<b>N</b>	4.88	1.10	5.96 <sup>#</sup>	1.80	2.75	4.0	4.0	4.0	0.05	1.0	0.0	2.00	2.80	0.00

	Binary system												
	$E_c$	$r_e$	$B$	$\rho_0$ (N/Ti)	$C_{min}$ (Ti,N,Ti )	$C_{min}$ (N,Ti,N)	$C_{min}$ (Ti,Ti,N )	$C_{min}$ (Ti,N,N)	$C_{max}$ (Ti,N,Ti )	$C_{max}$ (N,Ti,N)	$C_{max}$ (Ti,Ti,N )	$C_{max}$ (Ti,N,N)	$d$
TiN	6.615	2.121	320	18	0.16	0.09	0.81	1.46	2.80	2.80	1.44	2.80	0.00

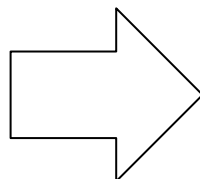
Units:  $E_c$  (eV),  $r_e$  (Å),  $B$  (GPa)

<sup>#</sup>alpha value

Reference structures: hcp (Ti), dimer (N), cubic-B1 (TiN)

# LAMMPS & MEAM

**LAMMPS MD Simulator**  
**Sandia National**  
**Laboratories**



- Large-scale atomic/molecular massively parallel simulator
- Open source code
- Many potentials for classical MD available

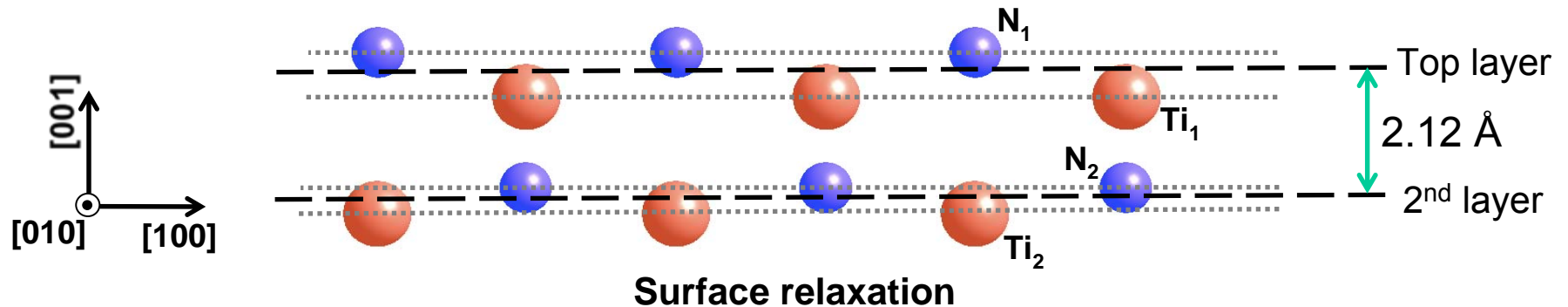
	TiN Bulk				
	MEAM Original work	Experimental <sup>1</sup>	DFT <sup>2</sup>	LAMMPS Original param.	LAMMPS Own param.
$E_c$ (eV)	-6.615			-6.608	-6.611
$a$ (Å)	4.242	4.240	4.254	4.243	4.241
$B$ (GPa)	320	318	290	319	294
$C_{11}$ (GPa)	659	625	640	736	659
$C_{12}$ (GPa)	150	165	115	101	112
$C_{44}$ (GPa)	183	163	159	179	171

<sup>1</sup>Kim et al. J. App. Phys., **72** (1992)

<sup>2</sup>Sangiovanni et al. Phys. Rev. B, **81** (2010)

# TiN surface results

	MEAM Original work	DFT <sup>1,2</sup>	LAMMPS Original param.	LAMMPS Own param.
<b><math>E_{surf}</math> (100)</b>				
Not relaxed (meV/Å <sup>2</sup> )	111	105	95	105
Relaxed (meV/Å <sup>2</sup> )	81	86	67	84
<b><math>E_{surf}</math> (111)</b>				
Not relaxed (meV/Å <sup>2</sup> )	245	311	232	236
Relaxed (meV/Å <sup>2</sup> )	228	226	211	217
<b>Surf Relaxation (001)</b>				
$\Delta$ dist. N <sub>1</sub> -Ti <sub>2</sub> (%)	7.7	2.1	5.7	5.3
$\Delta$ dist. Ti <sub>1</sub> -N <sub>2</sub> (%)	-5.0	-6.3	-5.5	-5.1
z dist. N <sub>1</sub> -Ti <sub>1</sub> (Å)	0.26	0.18	0.24	0.21



<sup>1</sup>Marlo et al. Phys. Rev. B, **62** (2000)

<sup>2</sup>Dudiy et al. Phys. Rev. B, **69** (2004)

# Diffusion energy barriers, previous results

	Energy Barrier (eV)
	Experimental <sup>1,2</sup>
<b><i>TiN (100) Surface</i></b>	
N adatom	1.1 - 1.4*
Ti adatom	1.1 - 1.4*
Ti-N <sub>x</sub> admolecules	1.1 - 1.4*
<b><i>TiN (111) Surface</i></b>	
N adatom	-
Ti adatom	-
Ti-N <sub>x</sub> admolecules	-

	Energy Barrier (eV)	
	DFT <sup>3</sup>	DFT <sup>4</sup>
<b><i>TiN (100) Surface</i></b>		
N adatom	0.95	0.95
Ti adatom	0.35	0.43
Ti-N dimer	>0.88	-
<b><i>TiN (111) Surface</i></b>		
N adatom	-	-
Ti adatom	1.74	-
Ti-N dimer	-	-

\* 1.1 eV for  $f_{N_2} = 0.1$ , 1.4 eV for  $f_{N_2} = 1.0$

<sup>1</sup> MA Wall et al., Phys. Rev. B, **70** (2004)

<sup>2</sup> MA Wall et al., Surf. Sci., **581** (2005)

<sup>3</sup> D Gall et al., J. App. Phys., **93** (2003)

<sup>4</sup> Present work, VASP, 15 x 15 x 1 **k** points, 5 layers of 16 atoms, PAW+GGA



# Diffusion energy barriers, original parameterization

	Experimental	DFT	DFT Present work	LAMMPS Original Param.
<i>N on TiN(001)</i> (eV)	1.1 - 1.4*	0.95	0.95	0.4
<i>Ti on TiN(001)</i> (eV)	1.1 - 1.4*	0.35	0.43	0.6

	Single element system													
	$E_c$	$r_e$	$B$	$A$	$\beta^{(0)}$	$\beta^{(1)}$	$\beta^{(2)}$	$\beta^{(3)}$	$\tau^{(1)}$	$\tau^{(2)}$	$\tau^{(3)}$	$C_{min}$	$C_{max}$	$d$
Ti	4.87	2.92	110	0.66	2.70	1.0	3.0	1.0	6.80	-2.0	-12.0	1.00	1.44	0.00
N	4.88	1.10	5.96 <sup>#</sup>	1.80	2.75	4.0	4.0	4.0	0.05	1.0	0.0	2.00	2.80	0.00

	Binary system												
	$E_c$	$r_e$	$B$	$\rho_0$ (N/Ti)	$C_{min}$ (Ti,N,Ti) )	$C_{min}$ (N,Ti,N)	$C_{min}$ (Ti,Ti,N) )	$C_{min}$ (Ti,N,N)	$C_{max}$ (Ti,N,Ti) )	$C_{max}$ (N,Ti,N)	$C_{max}$ (Ti,Ti,N) )	$C_{max}$ (Ti,N,N)	$d$
TiN	6.615	2.121	294	18	0.16	0.09	0.81	1.46	2.80	2.80	1.44	2.80	0.00

# Diffusion energy barriers, own parameterization

	Experimental	DFT	DFT Present work	LAMMPS Orig. Param.	LAMMPS Own Param.
<i>N on TiN(001)</i> (eV)	1.1 - 1.4	0.95	0.95	0.4	0.7
<i>Ti on TiN(001)</i> (eV)	1.1 - 1.4	0.35	0.43	0.6	0.9

- Present work DFT results agree with existing DFT values
- MEAM diffusion barriers higher than DFT values
- MEAM values considerably closer to experimental barriers
- Qualitative agreement with experimental findings – no preferred diffusing species on (001)

# Recommended CDIO project potential functionals

## 1. Lennard – Jones (LJ)

$$\Phi(r) = 4\varepsilon \left[ \left( \frac{\sigma}{r} \right)^{12} - \left( \frac{\sigma}{r} \right)^6 \right]$$

- well suited for inert gases (metals), mostly for gas-metals interactions.

### Values of LJ parameters

$\varepsilon$  [ $10^{-16}$  erg];  $\sigma$  [Å]

	He	Ne	Ar	Kr	Xe	Ni	Ag	Rh
$\varepsilon$	14	50	167	225	320	0.66 [eV]	0.34 [eV]	0.67 [eV]
$\sigma$	2.56	2.74	3.40	3.65	3.98	2.24	2.65	2.46

## 2. Morse potential

$$\Phi(r) = D \left( e^{-2\alpha(r-r_0)} - 2e^{-\alpha(r-r_0)} \right)$$

- D,  $r_0$ ,  $\alpha$  determined by fitting to reproduce bulk properties (lattice constant, cohesive energy, compressibility).
- used mostly for gas-metals (metals) interactions.

### Values of parameters for Morse potential

**D** [eV];  $r_0$  [Å];  $\alpha$  [Å<sup>-1</sup>]

	<b>D</b>	<b><math>r_0</math></b>	<b><math>\alpha</math></b>
<b>Ag</b>	0.325	3.13	1.36
<b>Rh</b>	0.665	2.87	1.54

## MD integrators

Selection criteria: *simplicity, efficiency, stability, reliability, parallelization.*

Most integrators are based on Taylor expansion of particle coordinates  $\mathbf{r}(t)$  around time, for small increments  $\Delta t$ , which is the MD time step.

### Euler Algorithm

$$\mathbf{r}(t + \Delta t) = \mathbf{r}(t) + \mathbf{v}(t)\Delta t + \frac{\mathbf{f}(t)}{2!m} \Delta t^2 + \frac{\Delta t^3}{3!} \ddot{\mathbf{r}} + O(\Delta t^4) + \dots$$

Algorithm is obtained by truncating the expansion beyond the  $\Delta t^2$  term.

$$\mathbf{r}(t + \Delta t) = \mathbf{r}(t) + \mathbf{v}(t)\Delta t + \frac{\mathbf{f}(t)}{2m} \Delta t^2$$

Similar to the Verlet algorithm but in fact much worse on all counts.

- it is not time-reversible
- introduces catastrophic energy drifts

Definitely NOT recommended.

## Verlet Algorithm (Verlet, Störmer, 1967)

$$r(t + \Delta t) = r(t) + v(t)\Delta t + \frac{f(t)}{2!m}\Delta t^2 + \frac{\Delta t^3}{3!}\ddot{r} + O(\Delta t^4) + \dots$$

$$r(t - \Delta t) = r(t) - v(t)\Delta t + \frac{f(t)}{2!m}\Delta t^2 - \frac{\Delta t^3}{3!}\ddot{r} + O(\Delta t^4) + \dots$$

Summing the above equations, one obtains:

$$r(t + \Delta t) + r(t - \Delta t) = 2r(t) + \frac{f(t)}{m}\Delta t^2 + O(\Delta t^4)$$

The equation for advancing the positions in Verlet algorithm becomes:

$$r(t + \Delta t) = 2r(t) - r(t - \Delta t) + \frac{f(t)}{m}\Delta t^2 \quad (1)$$

The errors in the estimations of new positions are of order  $\Delta t^4$ .

$r(t + \Delta t)$  and  $r(t - \Delta t)$  play symmetrical roles, the algorithm time-reversible.

Velocities are not required to compute trajectories.

Yet, velocities are needed to directly calculate kinetic energy and temperature.

Velocities can be computed from:

$$r(t + \Delta t) - r(t - \Delta t) = 2v(t)\Delta t + O(\Delta t^3)$$

and

$$v(t) = \frac{r(t + \Delta t) - r(t - \Delta t)}{2\Delta t} + O(\Delta t^2) \quad (2)$$

The errors in velocities estimations are of order  $\Delta t^2$ .

Though algorithm is stable, one can improve this accuracy.

### Leap-frog Algorithm (Hockney, 1970; Potter, 1972)

Derived from the Verlet algorithm, evaluates velocities at half-step:

$$v(t - \Delta t/2) = \frac{r(t) - r(t - \Delta t)}{\Delta t} \quad \text{and} \quad v(t + \Delta t/2) = \frac{r(t + \Delta t) - r(t)}{\Delta t}$$

New positions can be obtained as:

$$r(t + \Delta t) = r(t) + \Delta t v(t + \Delta t/2) \quad (3)$$

Velocities can be computed from:

$$v(t + \Delta t/2) = v(t - \Delta t/2) + \Delta t \frac{f(t)}{m} \quad (4)$$

Stored quantities are positions,  $r(t)$ , and accelerations,  $a(t) = f(t)/m$ .

Equation (4) is implemented first, and once  $v(t + \Delta t/2)$  are obtained, velocities  $v(t)$  at current time step are calculated to permit the computation of quantities which require  $r(t)$  and  $v(t)$ :

$$v(t) = \frac{1}{2} [v(t + \Delta t/2) + v(t - \Delta t/2)]$$

Upon completion of the half-step above, eq. (3) is used to calculate  $r(t + \Delta t)$ . Accelerations  $a(t)$  are also evaluated now and stored for next step.

Leap-from algorithm yields identical trajectories to those obtained with Verlet.

pros – significantly smaller numerical errors in velocities evaluation.

cons – positions and velocities are not evaluated at the same time step.



## Velocity Verlet Algorithm (Swope, anderson et. al. 1982)

$$r(t + \Delta t) = r(t) + v(t)\Delta t + \frac{f(t)}{2m} \Delta t^2 = r(t) + v(t)\Delta t + \frac{\Delta t^2}{2} a(t) \quad (5)$$

$$v(t + \Delta t) = v(t) + \frac{f(t) + f(t + \Delta t)}{2m} \Delta t = v(t) + \frac{\Delta t}{2} [a(t) + a(t + \Delta t)] \quad (6)$$

Implemented to store  $r(t)$ ,  $v(t)$  and  $a(t)$  **all** at the same time.

Similar to predictor-corrector algorithms, but equivalent to Verlet:

a: using (5) 
$$r(t + 2\Delta t) = r(t + \Delta t) + v(t + \Delta t)\Delta t + \frac{f(t + \Delta t)}{2m} \Delta t^2$$

b: from (5) 
$$r(t) = r(t + \Delta t) - v(t)\Delta t - \frac{f(t)}{2m} \Delta t^2$$

a+b = c 
$$r(t + 2\Delta t) + r(t) = 2r(t + \Delta t) + [v(t + \Delta t) - v(t)]\Delta t + \frac{f(t + \Delta t) - f(t)}{2m} \Delta t^2$$

(6) in c: 
$$r(t + 2\Delta t) + r(t) = 2r(t + \Delta t) + \frac{f(t + \Delta t)}{m} \Delta t^2 \equiv \text{positions in Verlet alg.}$$

## Velocity Verlet Algorithm

Requires storage of  $r(t)$ ,  $v(t)$ ,  $a(t)$ .

Consists of two stages and a force evaluation in between:

**Stage 1:**  $r(t + \Delta t)$  are evaluated using (5) and  $v(t + \Delta t/2)$  using:

$$v(t + \Delta t/2) = v(t) + \frac{\Delta t}{2} a(t)$$

**Intermediate stage:**  $f(t + \Delta t)$  and  $a(t + \Delta t)$  are computed.

**Stage 2:** velocity move is completed.

$$v(t + \Delta t) = v(t + \Delta t/2) + \frac{\Delta t}{2} a(t + \Delta t)$$

Attractive trade-off between numerical stability, convenience, simplicity.

Ensures excellent energy conservation, solid trajectories evolution.

Nearly universal choice in available MD codes, including QMD

### Beeman Algorithm (1976)

$$r(t + \Delta t) = r(t) + v(t)\Delta t + \frac{2}{3}a(t)\Delta t^2 - \frac{1}{6}a(t - \Delta t)\Delta t^2 \quad (7)$$

$$v(t + \Delta t) = v(t) + \frac{1}{3}a(t + \Delta t)\Delta t + \frac{5}{6}a(t)\Delta t - \frac{1}{6}a(t - \Delta t)\Delta t \quad (8)$$

This scheme requires storage of  $r(t)$ ,  $v(t)$ ,  $a(t)$  and  $a(t - \Delta t)$ .

Additional requirement to store  $a(t - \Delta t)$  compensated by improved energy conservation.

However, the algorithm is not time-reversible.

### Velocity-Corrected Verlet Algorithm

Errors in predicted positions and velocities are both of order  $\Delta t^4$ .

Based on Taylor expansions for  $r(t + 2\Delta t)$ ,  $r(t + \Delta t)$ ,  $r(t - \Delta t)$  and  $r(t - 2\Delta t)$

$$r(t + 2\Delta t) = r(t) + 2v(t)\Delta t + \dot{v}(t)(2\Delta t)^2/2! + \ddot{v}(t)(2\Delta t)^3/3! + \dots$$

$$r(t + \Delta t) = r(t) + v(t)\Delta t + \dot{v}(t)\Delta t^2 / 2! + \ddot{v}(t)\Delta t^3 / 3! + \dots$$

$$r(t - \Delta t) = r(t) - v(t)\Delta t + \dot{v}(t)\Delta t^2 / 2! - \ddot{v}(t)\Delta t^3 / 3! + \dots$$

$$r(t - 2\Delta t) = r(t) - 2v(t)\Delta t + \dot{v}(t)(2\Delta t)^2 / 2! - \ddot{v}(t)(2\Delta t)^3 / 3! + \dots$$

By combining the above equations one obtains:

$$12v(t)\Delta t = 8[r(t + \Delta t) - r(t - \Delta t)] - [r(t + 2\Delta t) - r(t - 2\Delta t)] + O(\Delta t^4)$$

or

$$v(t) = \frac{v(t + \Delta t/2) + v(t - \Delta t/2)}{2} + \frac{\Delta t}{12} [\dot{v}(t - \Delta t) - \dot{v}(t + \Delta t)] + O(\Delta t^4) \quad (9)$$

Significant improvement in errors but requires complex variables storage and calculations.

### Predictor-Corrector Algorithms

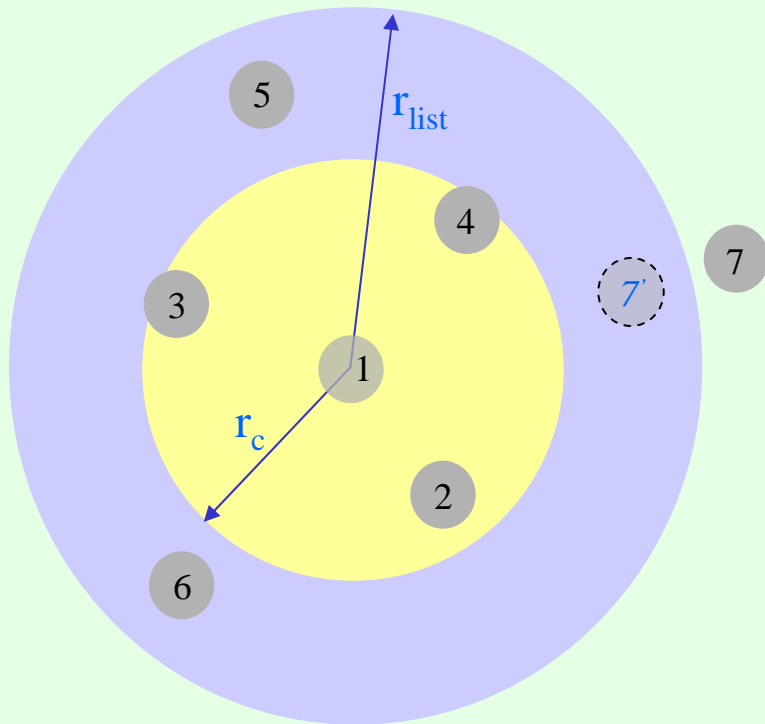
Several variants (Gear, etc), used mostly in conjunction with variable  $\Delta t$ .

Generally expensive in terms of CPU. More details in the book.

# Neighbors Lists

Use of PBCs, potential truncation and MIC still leaves  $O(N^2)$  pair interactions. Efficiency increases significantly if lists of neighbors are maintained & updated.

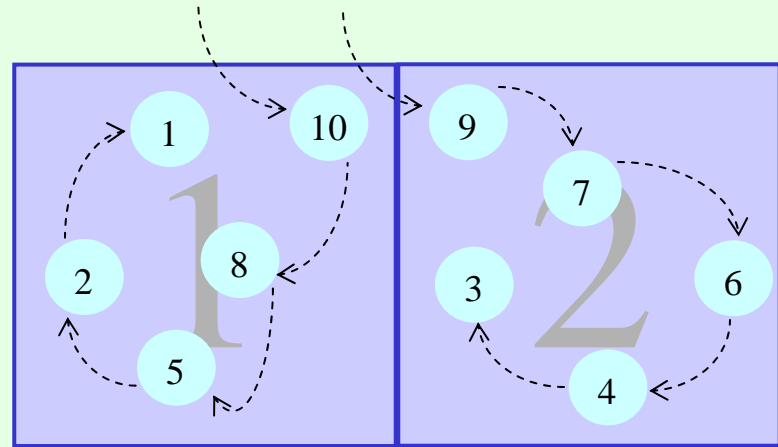
## Verlet List



- $r_{list}$  defines “*skin*” around cutoff sphere.
- only particles in *cutoff* & *skin* are kept.
- updating interval 10 – 100 time steps.
- larger *skin*, fewer updates, less efficient.
- method can reduce CPU time with 50%.
- one needs good balance between the size of the *skin* and updating interval.
- final choice dictated by studied problem, or phenomena.

## Cells structures and linked cells

31	32	33	34	35	36
25	26	27	28	29	30
19	20	21	22	23	24
13	14	15	16	17	18
7	8	9	10	11	12
1	2	3	4	5	6



- primary cell divided in  $M \times M \times M$ , each cell has  $N_c = N/M^3$  (in 3-dim).
- size of cells must be greater than cutoff distance.
- atoms in cell 14 only interact with those in cells 7, 8, 9, 13, 15, 19, 20 and 21.
- PBCs still apply to particles in cells 7, 13 and 19.
- two lists are made: HEAD-of-chain (cell id), and LIST of cell-head interactions.
- cells are linked. In can MD easily avoids double counting of  $ij$  interaction.
- number of interactions is reduced from  $N^2$  (no MIC) and  $1/2N(N-1)$  to  $27NN_c$  in MC and  $13.5NN_c$  in MD simulations.

## Long range corrections

Can be added to compensate for potential truncation. For Lennard-Jones,

$$V^{\text{LJ}}(r) = 4\epsilon \left[ \left( \frac{\sigma}{r} \right)^{12} - \left( \frac{\sigma}{r} \right)^6 \right] \text{ the average potential energy for any given atom } i \text{ is:}$$

$$V_i(r) = \frac{1}{2} \int_0^\infty 4\pi r^2 \rho(r) V(r) dr, \text{ where } \rho(r) \text{ - average number density at } r \text{ from } i.$$

Tail contributions can be estimated as (if for  $r \leq r_c$ ,  $\rho(r) \equiv \rho$ ):

$$V^{\text{tail}} = \frac{1}{2} \int_{r_c}^\infty 4\pi r^2 \rho(r) V(r) dr = \frac{1}{2} 4\pi \rho \int_{r_c}^\infty r^2 V(r) dr = \frac{8}{3} \pi \rho \epsilon \sigma^3 \left[ \frac{1}{1} \left( \frac{\sigma}{r_c} \right)^9 - \left( \frac{\sigma}{r_c} \right)^6 \right]$$

## Reduced Units

Quite useful when dealing with systems containing one type of atoms.

Unit of mass =  $m$ , the mass of atoms in systems

For LJ potential: unit of length =  $\sigma$ ; unit of energy =  $\varepsilon$ .

density:  $\rho^* = \rho\sigma^3$

temperature:  $T^* = k_B T / \varepsilon$

energy:  $E^* = E / \varepsilon$

pressure:  $P^* = P\sigma^3 / \varepsilon$

time:  $t^* = (\varepsilon / m\sigma^3)^{1/2} t$

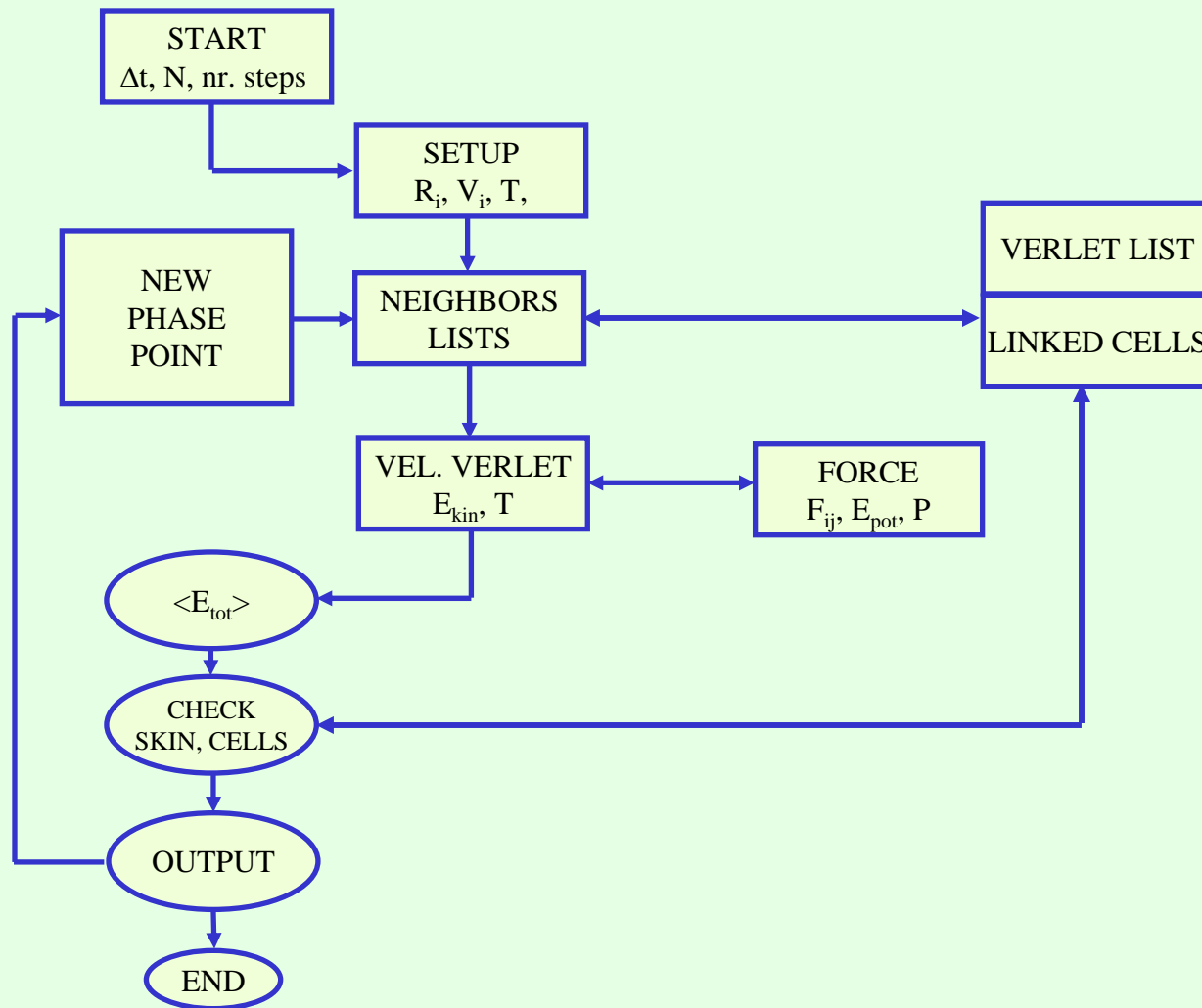
force:  $F^* = F\sigma / \varepsilon$

$$V^{*LJ}(r) = 4 \left[ \left( \frac{1}{r^*} \right)^{12} - \left( \frac{1}{r^*} \right)^6 \right]$$

**OBS: Whatever units are used,  
one must be consistent!  
Don't mix eV with m, N with Å.**



# Molecular Dynamics Flow Diagram



# How to setup MD

## START

- Define MD box (X, Y, Z), density, MD step & number of steps (total time).
- Type of simulation: new **AND** old (useful for continuation runs).
- Starting temperature and potential parameters (sometimes useful here).
- Flags for storage of calculated physical quantities (required for time averages).
- Flags for specific requirements in MD runs:
  - thermostat or NO thermostat
  - heat sink or NO heat sink
  - interval for storage of coordinates (useful for visualization, movies)
- Flags for data OUTPUT (required at regular intervals but not as often as for visualization).

REMEMBER: IN/OUT statements slow down code execution!

## INITIAL PHASE POINT

- Generate new AND/OR read existing atomic configurations ( $R_i$ ,  $V_i$  ).
- Provide for option to rescale velocities (required for simulations at new T).
- Calculate AND zero the total momentum (recommended).

## DO THIS BEFORE THE MD LOOP

- Potential tabulation (if used) and potential parameters.
- Define heat sink parameters (number of fixed layers, rule for energy dissipation)
- Define the thermostat method/algorithm.
- Define (if used) any external action (eg. electric field, stress) on the system.

## IMAGES/NEIGHBOURS LISTS

- Here one generates and stores the list of interactions to be used in the MD loop.

**For good efficiency one must:**

- Define and use Periodic Boundary Conditions (PBC)
  - **3-Dim** PBCs for **BULK** simulations
  - **2-Dim** PBCs for **SURFACE** simulations
- Provide for rapid counting of atoms (use Minimum Image Convention, Cells).
- Use potential truncation & shifting (based on potential range).
- **MAKE** and **STORE** the neighbors list (all valid PAIRS and TRIPLETS).
- Provide for possibility to update the list any time (every step OR every N steps).