

# MAE 298 Introduction to PDEs

## Electrochemical Modeling of Batteries

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### I. INTRODUCTION

**P**ARTIAL differential equations (PDEs) provide a mathematical basis for describing many significant scientific phenomena; however, while an important domain of study, in actual application it is often more challenging to understand and develop the PDE(s) that govern viable model than it is to understand the underlying mathematical formalism of PDEs. The inherent complexity of deriving and analyzing PDE models when compared to more approachable ODE models can be further exacerbated by systems of coupled equations and/or nonlinear dynamics. For these systems, the derivation and interpretation of the model are often obfuscated by modeling assumptions, domain specific knowledge, and an understanding of how these mechanisms are translated into a PDE formation. In effect, these PDE models compromise higher model accuracy at the expense of ease of modeling, computational overhead, intuition, and interpretability.

An important modern day example can be found in the study of electrochemical battery modeling, where even simplified electrochemical models consist of a system of partial differential equations, each of which describes a certain physical law which the dynamics of the battery must obey. The objective of this paper is to derive the salient PDEs that define the Single Particle Model (SPM) which is reasonable simplification of the well known Doyle-Fuller-Newman (DFN) model. The intent of this paper is to provide the reader with enough context that the reader will be able to understand the development of state of the art electrochemical battery modeling from fundamental assumptions to the principle mechanisms at work with this model, to inform how these concepts are translated into the mathematical language of Partial Differential Equations.

### II. SINGLE PARTICLE MODEL OVERVIEW

As alluded to above, the Doyle-Fuller-Newman model has long been held as the gold standard electrochemical battery model. This model describes at the first principle dynamics of a planar battery cell [Fig. 1]. This model encapsulates the dynamics of a lithium-ion battery at the continuum-scale. This is accomplished by deriving the microscale governing equations and applying volume averages over the models' domain. The primary elements of this model consist of five governing equations for the conservation of mass in homogeneous solid phase, the conservation of charge in homogeneous solid phase, the conservation of mass in homogeneous electrolyte phase, the conservation of charge in

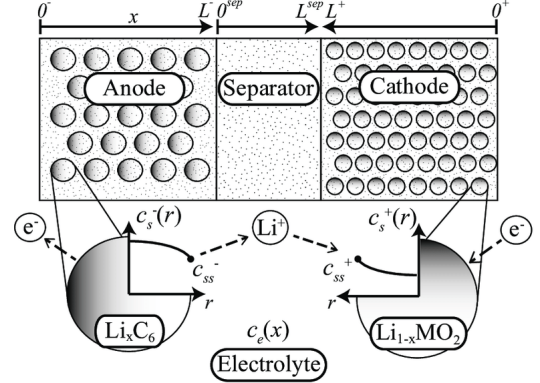


Fig. 1. Planar DFN Model Schematic

the homogeneous electrolyte phase, and finally the transport effects of lithium movement between the solid and electrolyte phases, known as the Butler-Volmer kinetics equation.

Unfortunately, derivation of the full DFN model would require investigation into the process of volume averaging which, in and of itself, is an involved procedure. To this end, this paper will endeavour to derive the Single Particle Model (SPM) which utilizes the same microscale governing equations as the DFN model. However, instead of applying volume averaging, SPM utilizes a reduction of order to simplify the battery model into a two dimensional problem of a single dimension of space and of time. SPM along with its' more accurate cousin (the Single Particle Model with Electrolyte Dynamics aka SPMe model), take the approach that simplifying the domain which the PDE defines, is an acceptable tradeoff of accuracy for a measurable boost in computational efficiency. As such these models have found growing acceptance in the domain of real-time or near real-time applications such as battery state of charge estimation, which has been dominated by lumped parameter ODE equivalent circuit models (ECM).

The majority of this paper will investigate the process of deriving the microscale governing partial differential equations, from which we can derive the Single Particle battery model. For reference the desired microscale model equations are listed below. Throughout this paper, the reader will see how individual components of physics, chemistry, and thermodynamics are brought together to derive the following set of partial differential equations.

$$\nabla \cdot \mathbf{i}_s = \nabla \cdot (-\sigma \nabla \phi_s) = 0 \quad (1)$$

$$\frac{\partial c_s}{\partial t} = \nabla \cdot (D_s \nabla c_s) \quad (2)$$

$$\frac{\partial c_e}{\partial t} = \nabla \cdot (D_e \nabla c_e) - \frac{\mathbf{i}_e \cdot \nabla t_+^0}{F} - \nabla \cdot (c_e \mathbf{v}_0) \quad (3)$$

$$\nabla \cdot \left( -\kappa \nabla \phi_e - \frac{2\kappa RT}{F} \left( 1 + \frac{\partial \ln f_{\pm}}{\partial \ln c_e} \right) (t_+^0 - 1) \nabla \ln c_e \right) = 0 \quad (4)$$

$$j = \frac{i_0}{F} \left\{ \exp \left( \frac{(1-\alpha)F}{RT} \eta \right) - \exp \left( -\frac{\alpha F}{RT} \eta \right) \right\} \quad (5)$$

### III. MICROSCALE MODEL DERIVATION

In the microscale model, the principle mechanisms of operation are evaluated under the assumption of homogeneous solid and electrolyte phases. With these assumptions, the model does not need to account for individual molecular interactions or impurities, as these are assumed to be averaged out to create a homogeneous material that can be directly used to derive and evaluate the governing equations in both the solid and electrolyte phases of the battery.

At the microscale, we are interested in deriving the governing equations which are pertinent to the operation of a lithium-ion battery. These governing equations are ...

- 1) Conservation of Mass in homogeneous solid phase
- 2) Conservation of Charge in homogeneous solid phase
- 3) Conservation of Mass in homogeneous electrolyte
- 4) Conservation of Charge in homogeneous electrolyte
- 5) Lithium transport effects

The two domains that we are interested in solving these equations for are the homogeneous solid which is represented by the battery electrodes, as well as the homogeneous electrolyte which represents the electrically conductive medium inbetween the anode and the cathode. When coupled with the dynamics of lithium transport into the active material of the electrode, these equations will define all of the salient dynamics which a lithium-ion battery will incorporate. We will begin with the conservation of charge in the solid phase.

#### A. Charge Conservation in Homogeneous Solid

From physics, we know that electric charge must be conserved in an isolated system. For the purposes of this paper, we will assume that the battery we are modeling constitutes an isolated system that does not interact electrically with its environment. This is a reasonable assumption. Therefore, if the conservation of charge must hold, then it must be true that charge is not created or destroyed inside of the solid electrodes. This is expressed symbolically as the following surface integral of current flux density.

$$I = - \oint_S \mathbf{i} \cdot d\mathbf{S} \quad (6)$$

Where the net current into an arbitrary surface  $\mathbf{S}$  can be written in terms of the surface integral of current density  $\mathbf{i}$ . Via application of the Divergence Theorem and Conservation of Charge, it can be shown that ...

$$I = - \iiint_V (\nabla \cdot \mathbf{i}) dV = 0 \quad (7)$$

Since we assume charge in the volume of integration must be conserved, and since current is the time rate of change of charge, we can state that the net current flow into/out of the volume of integration must be zero. In this form, we note that since the entire volume integral must equate to zero, the integrand must also be indentially equal to zero.

$$\nabla \cdot \mathbf{i} = 0 \quad (8)$$

While this statement is an accurate formulation of **conservation of charge**, we would like to formulate this same expressions as a function of electric potential, instead of current density. This can be done by applying the point form of **Ohm's Law**, namely  $\mathbf{i} = \sigma \mathbf{E}$ , where the electric field  $\mathbf{E} = -\nabla \phi$ .

$$\mathbf{i} = -\sigma \nabla \phi \quad (9)$$

By substituting equation (9) into (8), we arrive at the final form of the conservation of charge in the electrodes, shown below.

$$\nabla \cdot (-\sigma \nabla \phi_s) = 0 \quad (10)$$

When stated directly, equation (10) states that the divergence of the conductivity ( $\sigma$ ) weighted gradient of the electrical potential must equal zero for charge to be conserved in the volume of system which we have defined (aka the battery). This expression must hold true for both electrodes of the battery.

#### B. Mass Conservation in Homogeneous Solid

Next, the **conservation of mass** in homogeneous solid (electrodes) is used to capture the dynamics of lithium transport. The principle mechanism here is diffusion. As such, we can model this behavior as a diffusion process via Fick's first law, where the molar flux density  $\mathbf{N}$  [ $\text{mol m}^{-2} \text{s}^{-1}$ ] is related to the gradient of lithium concentration in the solid electrode material, proportional to a material dependent constant of diffusivity  $D$ .

$$\mathbf{N} = -D \nabla c \quad (11)$$

While equation (11) encapsulates the mechanism of diffusion of molar lithium as function of the concentration gradient, we would like to express it in terms of the time rate of change of the solid phase concentration. To accomplish this, we first need to compute the net molar flux by taking the integral of the net molar flux density over the entire domain boundary.

$$j = - \oint_S \mathbf{N} \cdot \hat{\mathbf{n}} dS \quad (12)$$

This provides us with the net molar flux  $\mathbf{j}$  [ $mol s^{-1}$ ], which can be rewritten in terms of a molar flow rate, where  $n$  is the number of moles of lithium ...

$$j = \frac{dn}{dt} \quad (13)$$

Recalling that the definition of molar concentration is the number of moles of a material over the total volume it occupies ( $c = n/V$ ), we can define the an infinitesimal mole  $dn$  to be computed by integrating each contribution of concentration within a differential volume over the entire volume of the domain and then taking the derivate with respect to time.

$$n = \iiint_V c dV \quad (14)$$

$$\frac{dn}{dt} = -\frac{d}{dt} \iiint_V c dV \quad (15)$$

By substituting equations (13) and (15) into equation (12) we arrive at the following statement.

$$\iint_S \mathbf{N} \cdot \hat{n} dS = -\frac{d}{dt} \iiint_V c dV \quad (16)$$

This is the integral form of the continuity equation, which relates the molar flux density, from Fick's First Law, to the concentration of lithium in the solid phase of the battery.

This is an important formulation because it provides the link to express the molar flux density in terms of lithium concentration. While initially a rather an opaque connection, in the study of electrochemical battery models, the ability to express find desired dynamics as a function of concentration is desirable since many of the quantities we are interested in understanding with this model are defined as a function of concentration. The best example of this being a batteries state of charge (SOC).

Additionally, as a continuity equation, it states that no mass has been created or destroyed inside the domain, further implying that all mass transfer happens only at the boundaries of the domain. Equation (16) can be further simplified into the more familiar point form of Fick's second law, as shown below.

$$\frac{\partial c_S}{\partial t} = \nabla \cdot (D_S \nabla c_S) \quad (17)$$

This law states that the total lithium flux density (in a given volume) must be equal to the rate of change of concentration within that volume.

### C. Charge Conservation in Homogeneous Electrolyte

When discussing the conservation of charge inside of the electrolyte phase of the battery, a non-trivial amount of thermodynamics and physical chemistry are required to fully contextualize and provide the groundwork which needs to be laid, to understand the principles for deriving the governing equations.

This section will endeavour to lay the foundation for the elements of **Concentrated Solution Theory**, which are required to formulate the basic tenets of charge conservation in non-trivial chemical solutions, such as a batteries electrolyte.

1) *Gibbs Free Energy*: A simple definition of Gibbs Free Energy is total energy required to create the system (with internal energy  $u$ ), plus the amount which boundary work ( $pV$ ) required to make room for the system, minus the energy which is obtained from the surrounding environment (TS), which has a temperate of  $T$ . Conceptually, the Gibbs Free Energy is the amount of useful energy which can be harnessed to do work in the system.

$$G = U + pV - TS \quad (18)$$

The **free energy** portion of the name, is important because it informs us of how much energy can be extracted from the system, so long as the temperature and pressure are held constant. Additionally, the sign of the Gibbs free energy dictates the direction that a chemical reaction will spontaneously take.

In the context of batteries, Gibbs free energy is used in the formulation of electrochemical potentials, which are discussed below.

2) *Molarity & Molality*: Since an electrolyte is a solution, it can be described as an intensive property via its molarity and molality, depending on which attributes of the solution are of interest. The molarity of a solution is defined to be the number of moles of solute per liter of solution.

$$c_i = \frac{n_{solute_i}}{V_{solution}} \quad (19)$$

Where as the molality of the solution, is defined to be the total number of moles of solute over the mass of the solvent.

$$m_i = \frac{n_{solute_i}}{m_{solvent}} \quad (20)$$

3) *Electrochemical Potential*: Understanding the previous sections is important because it leads to a discussion of electrochemical potentials. Especially in multispecies solutions, the conversion from extensive properties to intensive properties is subtle. In order to properly define these intensive properties, it is neccessary to define partial molar quantities. This can be written using partial derivatives and describes how much the Gibbs free energy (extensive) changes with the addition of only species to the solution. This can be written...

$$\bar{\mu} = \left( \frac{\partial G}{\partial n_i} \right)_{T,p,n_j(j \neq i)} \quad (21)$$

where  $n_i$  is the number of moles of species  $i$ , while keeping the temperature, pressure, and moles of other species constant. Given this definition, note that electrochemical potential is an intensive property which has been normalized via the species mass (in moles).

As can be seen by substituting equation 18 into equation 21, the total potential is comprised of several components, including internal and external potentials. This distinction between internal and external potentials can be written as...

$$\bar{\mu}_i = \bar{\mu}_{i,internal} + \bar{\mu}_{i,external} \quad (22)$$

In the absence of other potentials and forces from external fields, the effect of chemical and electrochemical potentials can be written as ...

$$\bar{\mu}_i = \mu_i + z_i F \phi \quad (23)$$

Where the  $z_i$  is the charge number,  $F$  is Faraday's constant, and  $\phi$  is the local electrostatic potential. This equation neatly separates the potential contributions from both chemical and electrical mechanisms. After applying Euler's theorem for homogeneous functions we can show that.

$$G(\mathbf{n}) = \mathbf{n} \cdot \nabla G(\mathbf{n}) = \sum_i n_i \frac{\partial G}{\partial n_i} \quad (24)$$

$$G = \sum_i n_i \bar{\mu}_i \quad (25)$$

This provides a very concise form for relating the Gibbs free energy of our battery with the total potentials (chemical & electrical) of the battery.

4) *Gibbs-Duhem Equation*: The **Gibbs-Duhem** equation provides an important relationship between electrochemical potentials, which reduces the number of simultaneous equations which we must solve. It's derivation begins with the Gibbs Free Energy as shown below.

$$G = U + pV - TS \quad (26)$$

$$dG = dU + d(pV) - d(TS) \quad (27)$$

$$= dq + dw + pdV + VdP - TdS - SdT \quad (28)$$

Note that so far, all that has been accomplished has been applying a derivate and expanding the expression via the product rule of differentiation. From thermodynamics we know  $dw = -pdV$  and that under the assumption of reversibility  $dq = TdS$ . After applying these substitutions, we arrive at the simplified expressions below.

$$dG = Vdp - SdT \quad (29)$$

We can use equations 21 and 25 to write the previous equation in terms of the total potential.

$$\sum_{i=1}^r n_i d\bar{\mu}_i = Vdp - SdT \quad (30)$$

By rearranging the equations we get ...

$$\sum_{i=1}^r n_i d\bar{\mu}_i - Vdp + SdT = 0 \quad (31)$$

If we assume that pressure and temperature are constant, then derivatives of  $p$  and  $T$  are both zero. Since we expect to operate the battery in a constant temperature and pressure environment, these are meaningful simplifications which yield the following expression for a system with  $i$  chemical species.

$$\sum_{i=1}^r n_i d\bar{\mu}_i = 0 \quad (32)$$

In the special case we are interested in (e.g a binary electrolyte) containing only two species, this equation can be explicitly written as

$$n_1 d\bar{\mu}_1 + n_2 d\bar{\mu}_2 = 0 \quad (33)$$

This is an important equation because it directly governs the relationship between the two potentials. If one is increased, the other must decrease to maintain the equality. We can leverage this relationship to eliminate an equation, in future calculations, since we can write either quantity in terms of the other. This equation can also be written using the concentration of each species by dividing by the total volume.

$$\sum_{i=1}^r c_i d\bar{\mu}_i = 0 \quad (34)$$

5) *Activity*: Unlike non-ionic solutions, ionic solutions do not distribute their ions randomly, this is because the electrical charge for either a cation or anion are preferentially oriented and attracted to ions of the opposite charge. A consequence of this behavior is that movement of ions in solution are artificially slow, due to the movement of a specific ion also attempting to drag the other ions which are in its immediate neighborhood in the solution.

Because of this phenomenon, we define **activity** of a species as an effective concentration, which accounts for this discrepancy.

$$a_i = c_i f_i \quad (35)$$

In the context of lithium-ion batteries, the concept of activity is important because the transport medium between each electrode is a neutral ionic solution, which has a fixed number of positive and negative charge carries, that facilitate transfer of charge inside the battery. Therefore modeling this non-intuitive behavior is an important step in capturing the complete dynamics at play.

6) *Stoichiometric Coefficient  $\nu$*  : In a balanced chemical equation, the total amount of matter (of each element/molecule) and the total amount of charge are conserved. This fact holds true for electrolytes. In the case of an electrolyte, the solution is comprised of solute dispersed throughout a solvent such that the overall solution is charge-neutral. An example is shown below.



In this example, the coefficients in front of each ion on the right-hand-side of the equation, are known as **stoichiometric**

**coefficients**, and are denoted with the symbol  $\nu$ . In the equation shown above, the stoichiometric coefficients are  $\nu_{Na^+} = 2$  and  $\nu_{SO_4^{2-}} = 1$ .

7) **Charge Number** : Similar to the stoichiometric coefficients, the **charge number** is also displayed in the chemical equation, as the super-script for each compound on the right-hand-side of the the balanced chemical equation. The charge number tells us how many excess electrons or protons exist in each given ion compound. The charge numbers for the equation above are  $z_{Na^+} = 1$  and  $z_{SO_4^{2-}} = -2$ . The sign of the charge number informs us whether each compound is a cation ( $z > 0$ ) or anion ( $z < 0$ )

8) **Electroneutrality in Binary Electrolytes** : As mentioned previously electroneutrality is an important and frequently referenced concept in the development of the equations. In particular, a binary electrolyte comprised of only species is of particular interest and has useful properties which are very relevant to lithium-ion battery models. Under the constraint of charge-neutrality, we can state that the net charge in the solution must be  $q = 0$ . This can be expressed via charge numbers and the stoichiometric coefficients described above as follows.

$$\sum_i z_i v_i = 0 \quad (37)$$

For the general case the subscripts indicated whether the charge is from a cation or an anion, or the contributions from the solvent.

$$z_+ v_+ + z_- v_- + z_0 v_0 = 0 \quad (38)$$

However, since the solvents in the electrolyte are always charge neutral, we can omit them from this equation, leaving the following expression.

$$z_+ v_+ + z_- v_- = 0 \quad (39)$$

$$\frac{c_+}{v_+} = \frac{c_-}{v_-} \quad (40)$$

This can be translated from stoichiometric coefficients to the ion concentration of each species, through the fact that we know these quantities are proportional to each other where the constant of proportionality is the concentration of the solution in general.

$$c = \frac{c_+}{\nu_+} = \frac{c_-}{\nu_-} \quad (41)$$

We can make use of this attribute and rewrite the charge neutrality equation as follows.

$$\sum_i z_i c_i = 0 \quad (42)$$

Where  $i = 2$  for our use-case of a binary electrolyte. This expression will be useful later when deriving the conservation of charge in an electrolyte.

9) **Current Flow in Electrolyte** : Current flow inside of the electrolyte is comprised of both cations and anions. Therefore the total current flow inside of the electrolyte is given by

$$i = i_+ + i_- \quad (43)$$

The next step is try write these ionic currents in terms of flux density so that they can be written in standard form later. This is accomplished by accounting for the amount of charge per second (for both cations and anions), due to the respective flux.

$$i_+ = z_+ F N_+ = z_+ F c_+ v_+ \quad (44)$$

$$i_- = z_- F N_- = z_- F c_- v_- \quad (45)$$

Upon reflection of the quantities used in this expression we can see that the product of charge number, Faraday's constant, and flux density, is the electrical current produced for the given charge type (positive/negative).

Therefore,

$$i = z_+ F c_+ v_+ + z_- F N_- = z_- F c_- v_- \quad (46)$$

$$= F \sum_i z_i N_i \quad (47)$$

10) **Deriving Conservation of Charge** : To begin this derivation, we can start off with the general expression for the conservation of mass at any point in a solution.

$$\frac{\partial c_i}{\partial t} = -\nabla \cdot \mathbf{N}_i + R_i \quad (48)$$

This version of Ficks Law states that the time rate of change on the concentration for species  $i$  is related to the negative flux density gradient of that species as well as the rate of generation of species  $i$ .

$$F \frac{\partial z_i c_i}{\partial t} = -\nabla \cdot (z_i F \mathbf{N}_i) + z_i F R_i \quad (49)$$

Via algebra, we can multiply each side of the equation by  $z_i F$  to get,

$$\frac{\partial}{\partial t} F \sum_i z_i c_i = -\nabla \cdot \left( F \sum_i z_i \mathbf{N}_i \right) + F \sum_i z_i R_i \quad (50)$$

Since the previous equation was written only with respect to a single species in the equation, we need to sum the same equation over each species resulting in the following...

$$\frac{\partial}{\partial t} F \sum_i z_i c_i = -\nabla \cdot \left( F \sum_i z_i \mathbf{N}_i \right) \quad (51)$$

As mentioned in the previous section about charge neutrality, we can see that our previous equation that the summation  $\sum_i z_i c_i$  must be identically equal to zero since, the overall charge of the electrolyte is zero. We can also see that the term  $F \sum_i z_i \mathbf{N}_i$  matches our previous definition for current shown

above. By substituting these terms into the previous expression we arrive at the following equation.

$$\nabla \cdot \mathbf{i} = 0 \quad (52)$$

This equation is the differential form for the conservation of charge inside the electrolyte, stating that charge and not be created or destroyed.

#### D. Mass Conservation in Homogeneous Electrolyte

This next section is one of the most challenging to conceptualize as it has a great deal of nuance and complexity.

1) *Maxwell-Stefan Relationship*: This section lays the mathematical groundwork for the interaction of particles of differing species and the net force associated with these collisions. By applying the conservation of momentum, we can define the following expression for using average species velocities, for a 2 species mixture.

$$m_1 \mathbf{v}_{m_1} + m_2 \mathbf{v}_{m_2} = m_1 \mathbf{v}'_{m_1} + m_2 \mathbf{v}'_{m_2} \quad (53)$$

Given that the change in momentum for a single species can be written as...

$$\Delta(m_1 \mathbf{v}_{m_1}) = m_1 (\mathbf{v}_{m_1} - \mathbf{v}'_{m_1}) \quad (54)$$

We can isolate the post-collision velocity of species 1, as follows.

$$\mathbf{v}'_1 = \frac{m_1 \mathbf{v}_1 + m_2 \mathbf{v}_2}{m_1 + m_2} \quad (55)$$

These two previous expressions can be combined to form the following equation.

$$\begin{aligned} \Delta(m_1 \mathbf{v}_1) &= m_1 (\mathbf{v}_1 - \mathbf{v}'_1) \\ &= m_1 \left( \mathbf{v}_1 - \frac{m_1 \mathbf{v}_1 + m_2 \mathbf{v}_2}{m_1 + m_2} \right) \end{aligned} \quad (56)$$

Which can be simplified to...

$$= \frac{m_1 m_2}{m_1 + m_2} (\mathbf{v}_1 - \mathbf{v}_2) \quad (57)$$

2) *Momentum Charge Rate*: Now the task is to relate the change of momentum equation (on a mass basis) to the equivalent form on a species concentration basis. Generally speaking we can state that as the concentrations of either species in a two species mixture increase the change in momentum should increase proportionally. Putting this intuition into math, we arrive at the following statement.

$$\left( \frac{dp}{dt} \right)_V^{\text{species-}} \propto c_1 c_2 (\mathbf{v}_1 - \mathbf{v}_2) \quad (58)$$

From this statement, we can use the definition of the mole fraction  $x_i = n_i/n$ , where  $n_i$  is the number of moles of a given species and  $n$  is the total number of moles in the system. This defines the following relationship  $x_1 + x_2 = 1$  for a binary species and in general form can be written as

$\sum_i x_i = 1$ . Furthermore, we can define a quantity called the total concentration, such that.

$$C_T = \sum_i c_i \quad (59)$$

Where concentration is defined as  $c = n/V$ . From this relationship we can show that.

$$x_i = \frac{n_i}{n} = \frac{n_i/V}{n/V} = \frac{c_i}{C_T} \quad (60)$$

Which can be substituted directly into the change of momentum equation to produce...

$$\left( \frac{dp}{dt} \right)_V^{\text{species-}} \propto x_1 x_2 (\mathbf{v}_1 - \mathbf{v}_2) \quad (61)$$

From undergraduate physics we know from Newton's Second law that force is equivalent to the time rate of change of momentum, leading us to the following statement.

$$F_{1,V} \propto x_1 x_2 (\mathbf{v}_1 - \mathbf{v}_2) \quad (62)$$

In order to turn this proportionality into an equality we must define a constant of proportionality. Given the current form of this statement relating force to a type of relative velocity, we could guess that this constant would behavior very closely to a coefficient of friction,  $K_{ij}$ . In addition to this, we would like to break down this parameter, since it in itself is not a constant number since its value depends on that mole fraction and hence the concentration of the species in the system.

$$K_{ij} \propto x_1 x_2 \quad (63)$$

It turns out that the resulting equality has been defined in terms of the Maxwell-Stefan diffusivity  $D$ , and the surrounding pressure.

$$\mathcal{D}_{ij} = \frac{x_i x_j p}{K_{ij}} \quad (64)$$

By applying the ideal gas law and substituting the concentration form of the mole fraction developed earlier, we get.

$$K_{ij} = \frac{RT c_i c_j}{C_T \mathcal{D}_{ij}} \quad (65)$$

After developing these expressions, we can substitute them back into the original equation for the force acting on species 1.

$$\mathbf{F}_{1,V} = \frac{RT c_1 c_2}{C_T \mathcal{D}_{12}} (\mathbf{v}_1 - \mathbf{v}_2) \quad (66)$$

Now that we have an expression for the force acting on a species, we can begin to make the connections between forces and the electrochemical potential for charged species.

3) *Multicomponent Diffusion Equation:* We know that we can write force as the negative gradient of a potential function. In our case, this potential function is the Gibbs free energy developed previously. Furthermore, we can use the chain rule of calculus, to write the gradient of Gibbs free energy in terms of the gradient of the electrochemical potential and the rate of change of Gibbs free energy with respect to electrochemical potential. By reversing the relationship of Gibbs free energy, moles of a species, and the electrochemical potential developed earlier, it can also be shown that this last term is simply the number of moles of a species.

$$F_1 = -\nabla G_1 = -\frac{\partial G_1}{\partial \mu_1} \nabla \bar{\mu}_1 = -n_1 \nabla \bar{\mu}_1 \quad (67)$$

When this expression is normalized by volume, we can write the force of the particle in terms of species concentration.

$$F_{1,V} = \frac{F_1}{V} = -\frac{n_1}{V} \nabla \bar{\mu}_1 = -c_1 \nabla \bar{\mu}_1 \quad (68)$$

By equating Eqn (66) with the previous expression for force developed above, we get the following equation.

$$c_1 \nabla \mu_1 = RT \frac{c_1 c_2}{c_T \mathcal{D}_{12}} (\mathbf{v}_2 - \mathbf{v}_1) \quad (69)$$

This can be re-written...

$$c_i \nabla \mu_i = RT \sum_j \frac{c_i c_j}{c_T \mathcal{D}_{ij}} (\mathbf{v}_j - \mathbf{v}_i) = \sum_j K_{ij} (\mathbf{v}_j - \mathbf{v}_i) \quad (70)$$

After summing over all species in the mixture.

$$\sum_i c_i \nabla \mu_i = \sum_i \sum_j K_{ij} (\mathbf{v}_j - \mathbf{v}_i) \quad (71)$$

The significance of this equation is that it produces a basis for analysis use quantities available at the macroscopic level, and will be used later in the derivation of mass conservation in electrolyte.

#### 4) *Concentrated Binary Electrolyte Theory: Ion Fluxes:*

The objective of this next subsection, in the journey to arrive at an expression for conservation of mass in a binary electrolyte, is to define the ionic flux density,  $N_{\pm}$ .

The presence of ions in the electrolyte and the physical chemistry associated with an excess of charge in the solution, proves to be significantly more challenging and complex than deriving conservation of mass in a non-ionic medium. The three primary reasons for this are the caused by the chemical potential, average diffusivity, and the transference number for the electrolyte, which will each be addressed in turn before arriving at an expression for the flux-density.

Previously, we developed expressions for the electrochemical potentials for each species; however, for this analysis we would like to define a chemical potential for the electrolyte as a whole, instead of its constituent components, individually. This is accomplished by defining the electrolyte chemical potential as the stoichiometry weighted sum of the positive and negative electrochemical potentials, as follows.

$$\mu_e = v_+ \bar{\mu}_+ + v_- \bar{\mu}_- \quad (72)$$

This expression can then be expanded by Eqn (), which expands the electrochemical potential of the positive and negative ionic species.

$$\begin{aligned} \mu_e &= v_+ \bar{\mu}_+ + v_- \bar{\mu}_- \\ &= v_+ (\mu_+ + z_+ F \phi) + v_- (\mu_- + z_- F \phi) \\ &= v_+ \mu_+ + v_- \mu_- + (v_+ z_+ + v_- z_-) F \phi \\ &= v_+ \mu_+ + v_- \mu_- \end{aligned} \quad (73)$$

Note that in this expansion, the electrical potential will cancel out, since our battery is a binary electrolyte, that is overall charge neutral in solution. This means that the resulting potential is not a electrochemical potential, but only a chemical potential as the net electrical effect is zero.

Once we have the expression for the electrolyte chemical potential, it is a simple formality to find the gradient of this equation, and we can further multiple each

$$\nabla \mu_e = v_+ \nabla \bar{\mu}_+ + v_- \nabla \bar{\mu}_- \quad (74)$$

The reason for computing the gradient of the electrolyte chemical potential is to develop an expression which can be substituted into Eqn (71). As a reduction of notion, note that we will define  $c_{\pm}$  as shown below

$$c v_+ \nabla \bar{\mu}_+ = c_+ \nabla \bar{\mu}_+ \quad (75)$$

$$c v_- \nabla \bar{\mu}_- = c_- \nabla \bar{\mu}_- \quad (76)$$

By substituting the definition of the chemical potential gradient in Eqn(74), and the expressions we have for  $c_{\pm}$  shown above, into Eqn(71), we arrive at the following two equations.

$$\begin{aligned} c_+ \nabla \bar{\mu}_+ &= K_{+0} (\mathbf{v}_0 - \mathbf{v}_+) + K_{+-} (\mathbf{v}_- - \mathbf{v}_+) \\ c_- \nabla \bar{\mu}_- &= K_{-0} (\mathbf{v}_0 - \mathbf{v}_-) + K_{-+} (\mathbf{v}_+ - \mathbf{v}_-) \end{aligned} \quad (77)$$

After collecting like terms and canceling where valid, this equation can be written as ...

$$c \nabla \mu_e = K_{0+} (\mathbf{v}_0 - \mathbf{v}_+) + K_{0-} (\mathbf{v}_0 - \mathbf{v}_-). \quad (78)$$

Which will be used in subsequent part of the flux-density derivation. //

The second component which we would like to define for the electrolyte as a whole is the diffusion coefficient. To accomplish this, we would like to combine the effects of diffusion from both ionic species in the electrolyte. As with the chemical potential of the electrolyte, the diffusion coefficient will be defined as the stoichiometric weighted average of the diffusion coefficient for both of its constituent species. However, unlike the chemical potentials, diffusivity adds reciprocally.

$$\frac{v}{\mathcal{D}} = \frac{v_+}{\mathcal{D}_{0+}} + \frac{v_-}{\mathcal{D}_{0-}} \quad (79)$$

By solving for the diffusivity  $\mathcal{D}$  and multiplying by the concentration, as well as applying simplification from Eqns (75 & 76) we obtain...

$$\mathcal{D} = \frac{\nu C}{\frac{C_+}{\mathcal{D}_{0+}} + \frac{C_-}{\mathcal{D}_{0-}}} \quad (80)$$

If we further multiple this expressions by  $\frac{RTC_0}{C_T}$  over itself, we have only multiplied by unity and not changed the value of the expressions; however, after applying this transformation, this equation can be written in terms of the Maxwell-Stefan friction coefficients derived previously.

$$\mathcal{D} = \frac{vRTc_0c}{c_T(K_{0+} + K_{0-})} \quad (81)$$

The third and final component required to define the flux-density, is the transference numbers of the electrolyte. As mentioned previously the ionic nature of the electrolyte has the consequence of incorporating several non-intuitive chemistry tid-bits which require a more subtle analysis.

In simple terms the transference number represents the portion of the ionic current which is being carried by a specific species when there exists no gradient in the chemical potential.

Intuitively, it should make sense the transference number is propotional to amount of friction the ion will encounter, as such we can define this proportionality for both positive and negative species of the electrolyte.

$$t_+^0 \propto K_{0-} \quad \text{and} \quad t_-^0 \propto K_{0+} \quad (82)$$

In fact, for a binary electrolyte, we know that the total transference must be equal to unity, since either the positive or negative ionic current flow must be responsible for the net current. Given the previous proportionality, and the constraint, we can determine the constant of proportionality directly, which yields the following result.

$$t_+^0 = \frac{K_{0-}}{K_{0-} + K_{0+}} \quad \text{and} \quad t_-^0 = \frac{K_{0+}}{K_{0-} + K_{0+}} \quad (83)$$

#### PROVING THE ION-FLUX EQUATION:

Given this groundwork we can now begin to assemble the equation for flux-density. We first begin with the definition of flux-density

$$\mathbf{N}_i = c_i \mathbf{v}_i \quad \text{where } i = \pm \quad (84)$$

The intuition behind this expression is that the flux-density will change propotionally to the average velocity of a species and as a function of the concentration of the species, both of which make intuitive sense.

$$\mathbf{N}_+ = c_+ \mathbf{v}_+ = \underbrace{-\frac{v_+ \mathcal{D}}{vRT} \frac{c_T}{c_0}}_A \underbrace{\nabla \mu_e}_B + \underbrace{\frac{\mathbf{it}_+^0}{z_+ F}}_C + \underbrace{c_+ \mathbf{v}_0}_D \quad (85)$$

However, we still need an expression for the gradient of the chemical potential of the electrolyte.

Note that we can expand  $\nabla \mu_e$  via the chain rule of calculus, such that.

$$\nabla \mu_e = \frac{\partial \mu_e}{\partial c} \nabla c \quad (86)$$

And furthermore, we can take this expansion via concentration and use the chain rule again to write  $\nabla \mu_e$  in terms of the molality of the electrolyte, such that.

$$\frac{\partial \mu_e}{\partial c} = \frac{\partial \mu_e}{\partial \ln m} \frac{\partial \ln m}{\partial c} \quad (87)$$

Note that the natural-log is merely a build-in substitution that will enable a clear derivation in subsequent steps.

By using absolutely activity, we can show that...

$$\mu_i = RT \ln(\lambda_i) \quad (88)$$

$$\lambda_i = m_i \gamma_i \lambda_i^\ominus \quad (89)$$

By substituting these terms into the expression for chemical potential of the electrolyte ( $\mu_e = v_+ \mu_+ + v_- \mu_-$ ), and simplifying the resulting expression we get.

$$\begin{aligned} \mu_e &= vRT (\ln m + \ln \gamma_\pm) + \dots \\ &\quad v_+ RT \ln(v_+ \lambda_+^\ominus) + \dots \\ &\quad v_- RT \ln(v_- \lambda_-^\ominus) \end{aligned} \quad (90)$$

Such that we can define ...

$$\frac{\partial \mu_e}{\partial \ln m} = vRT \left( 1 + \frac{\partial \ln \gamma_\pm}{\partial \ln m} \right) \quad (91)$$

We can also define, the second term of of  $\nabla \mu_e$  as

$$m = \frac{m_+}{v_+} = \frac{c_+}{v_+ c_0 M_0} = \frac{c}{c_0 M_0} \quad (92)$$

$$\ln m = \ln c - \ln c_0 - \ln M_0 \quad (93)$$

$$\frac{\partial \ln m}{\partial \ln c} = 1 - \frac{\partial \ln c_0}{\partial \ln c} \quad (94)$$

$$\frac{\partial \ln m}{\partial c} = \frac{1}{c} \left( 1 - \frac{\partial \ln c_0}{\partial \ln c} \right) \quad (95)$$

$$\nabla \mu_e = \frac{vRT}{c} \left( 1 + \frac{d \ln \gamma_\pm}{d \ln m} \right) \left( 1 - \frac{d \ln c_0}{d \ln c} \right) \nabla c \quad (96)$$

This value can then be substituted back into the expression we derived for the flux-density.

$$\mathbf{N}_+ = -\frac{v_+ \mathcal{D}}{vRT} \frac{c_T}{c_0} c \nabla \mu_e + \frac{\mathbf{it}_+^0}{z_+ F} + c_+ \mathbf{v}_0 \quad (97)$$

$$D = \mathcal{D} \frac{c_T}{c_0} \left( 1 + \frac{d \ln \gamma_\pm}{d \ln m} \right) \quad (98)$$

$$\mathbf{N}_+ = -v_+ D \left( 1 - \frac{d \ln c_0}{d \ln c} \right) \nabla c + \frac{\mathbf{it}_+^0}{z_+ F} + c_+ \mathbf{v}_0 \quad (99)$$

$$\mathbf{N}_+ = -v_+ \frac{\mathcal{D} c_T}{c_0} \left( 1 + \frac{d \ln \gamma_\pm}{d \ln m} \right) \left( 1 - \frac{d \ln c_0}{d \ln c} \right) \nabla c + \frac{\mathbf{it}_+^0}{z_+ F} + c_+ \mathbf{v}_0 \quad (100)$$



Now that we have a full expression for the flux-density written fully in terms which are meaningful for us and our analysis, we can apply the Continuity equation as shown below, and then substitute in our flux-density equation.

$$\frac{\partial c_+}{\partial t} = -\nabla \cdot \mathbf{N}_+ \quad (101)$$

Which results in the following partial differential equation that enforces mass conservation inside of the electrolyte.

$$\frac{\partial c}{\partial t} = \nabla \cdot \left[ D \left( 1 - \frac{d \ln c_0}{d \ln c} \right) \nabla c \right] - \frac{\mathbf{i} \cdot \nabla t_+^0}{z_+ v_+ F} - \nabla \cdot (c \mathbf{v}_0) \quad (102)$$

#### E. Concentrated Solution Theory: Electrolyte Conservation

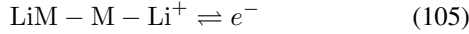
This section focuses on the conservation of charge inside of the binary electrolyte of the battery. This analysis will result in the derivation of the following equation.

$$\mathbf{i} = -\kappa \nabla \phi - \frac{\kappa}{F} \left( \frac{s_+}{n v_+} + \frac{t_+^0}{z_+ v_+} - \frac{s_0 c}{n c_0} \right) \nabla \mu_e \quad (103)$$

First, it is necessary to understand the chemical equation at play, and which must be balanced.



This can be written in an equivalent but more useful form as...



This equation represented interaction of lithium and some metal oxide which forms the active electrode material, as well as producing excess electrons. From physics it is known that the energy of this reaction must be conserved. Therefore, we can state that the Gibbs free energy of the reactants in the LHS of equation 105, must be equal to the electrical energy of the electrons produced on the RHS of the same equation.

$$\sum G = \sum E(e) \quad (106)$$

From previous sections we already know that we can write the Gibbs free energy and the energy due to the local electrostatic potential as follows.

$$\sum_i s_i \bar{\mu}_i = -nF\phi \quad (107)$$

By taking the gradient of this expressions we can derive the following relationship between the gradients of the electrochemical potential and the electrostatic potential.

$$\sum_i s_i \nabla \bar{\mu}_i = -nF \nabla \phi \quad (108)$$

By application of the Gibbs-Duhem equation 33, and accounting for the change from moles to concentration, we write the following expression for the electrolyte.

$$c_+ \nabla \bar{\mu}_+ + c_- \nabla \bar{\mu}_- + c_0 \nabla \bar{\mu}_0 = 0 \quad (109)$$

This expression encapsulates the positive, negative, and neutral components of the electrolyte solution, such that  $i = +, -, 0$ . Thus far, we have only encountered expressions for positive and negative electrochemical potentials but nothing for the neutral case. Thankfully, by leveraging the Gibbs-Duhem equation, 33 from previous sections, we can solve for the gradient of the 0th electrochemical potential. It can be shown that the expression can be written in terms of the electrolytes potential gradient, by substituting equation 74.

$$\begin{aligned} \nabla \bar{\mu}_0 &= -\frac{1}{c_0} (c_+ \nabla \bar{\mu}_+ + c_- \nabla \bar{\mu}_-) \\ &= -\frac{c}{c_0} (\nu_+ \nabla \bar{\mu}_+ + \nu_- \nabla \bar{\mu}_-) \\ &= -\frac{c}{c_0} \nabla \mu_e \end{aligned} \quad (110)$$

Via direct substitution, it is obvious that we can rewrite equation 108, in expanded form as follows.

$$s_+ \nabla \bar{\mu}_+ + s_- \nabla \bar{\mu}_- - s_0 \frac{c}{c_0} \nabla \mu_e = -nF \nabla \phi \quad (111)$$

From the balanced chemical equation, we can write.

$$s_+ z_+ + s_- z_- = -n \quad (112)$$

From this expression we can solve for  $s_-$  to arrive at the following equation.

$$s_- = -\left( \frac{z_+}{z_-} s_+ + \frac{n}{z_-} \right) \quad (113)$$

With this expression for  $s_-$  we can substitute it into component  $D$  conservation of energy equation, rewritten below.

$$\underbrace{s_+ \nabla \bar{\mu}_+ + s_- \nabla \bar{\mu}_-}_D - s_0 \frac{c}{c_0} \nabla \mu_e = -nF \nabla \phi \quad (114)$$

After applying this substitution, significant algebraic simplifications can be made to reduce the expression for term  $D$  to...

$$\underbrace{s_+ \nabla \bar{\mu}_+ + s_- \nabla \bar{\mu}_-}_D = \frac{s_+}{v_+} \nabla \mu_e - \frac{n}{z_-} \nabla \bar{\mu}_-. \quad (115)$$

This step has accomplished the task of rewriting the electrochemical gradients without the need to reference the positive electrode electrochemical potential gradient, and has reduced our original conservation of energy equation to ...

$$\begin{aligned} \frac{s_+}{v_+} \nabla \mu_e - \frac{n}{z_-} \nabla \bar{\mu}_- - s_0 \frac{c}{c_0} \nabla \mu_e &= -nF \nabla \phi \\ \left( \frac{s_+}{n v_+} - \frac{s_0 c}{n c_0} \right) \nabla \mu_e - \frac{1}{z_-} \nabla \bar{\mu}_- &= -F \nabla \phi \end{aligned} \quad (116)$$

However, we would still like express this term solely using  $\nabla \mu_e$ . Recall from the Maxwell-Stefan, equation, that we can write...

$$c_- \nabla \bar{\mu}_- = K_{0-} (\mathbf{v}_0 - \mathbf{v}_-) + K_{+-} (\mathbf{v}_+ - \mathbf{v}_-) \quad (117)$$

We can furthermore, use the equation for flux-density Eqn(118) and rearrange it such that...

$$c_i (\mathbf{v}_i - \mathbf{v}_0) = -\frac{v_i \mathcal{D}}{vRT} \frac{c_T}{c_0} \nabla \mu_e + \frac{\mathbf{i} t_i^0}{z_i F} \quad \text{for } i = \pm \quad (118)$$

Recall that  $\mathcal{D}$  merely represents the following collection of terms...

$$\mathcal{D} = \frac{vRT c_0 c}{c_T (K_{0+} + K_{0-})} \quad (119)$$

By applying the definition of the diffusion coefficient, dividing the equation by  $c_i$ , and simplifying that we can express both the positive and negative components as...

$$\begin{aligned} \mathbf{v}_+ - \mathbf{v}_0 &= -\frac{c}{K_{0+} + K_{0-}} \nabla \mu_e + \frac{\mathbf{i} t_+^0}{c_+ z_+ F} \\ \mathbf{v}_- - \mathbf{v}_0 &= -\frac{c}{K_{0+} + K_{0-}} \nabla \mu_e + \frac{\mathbf{i} t_-^0}{c_- z_- F} \end{aligned} \quad (120)$$

We can eliminate the term  $\mathbf{v}_0$  by subtracting the two equation above from each other, resulting in following equation, which has further been reduced by applying

$$\mathbf{v}_+ - \mathbf{v}_- = \frac{\mathbf{i}}{F} \left( \frac{t_+^0}{c_+ z_+} - \frac{t_-^0}{c_- z_-} \right) \quad (121)$$

This term can again be reduced. This is achieved by recalling that the sum of the transference number must equate to unity. In the case of a binary electrolyte, this implies that ...

$$t_+^0 + t_-^0 = 1 \quad (122)$$

By using this relationship and the definition of the transference number in Eqn(83), it is trivial to reduce the expression to the following equation.

$$\mathbf{v}_+ - \mathbf{v}_- = \frac{\mathbf{i}}{c_+ z_+ F} \quad (123)$$

This gives us the building blocks upon which we can use to write the conservation of energy using only  $\nabla \mu_e$ . If we start the Maxwell-Stefan equation again where  $i = -$ , divide both sides by  $c_- z_-$  that we will arrive at the following expression.

$$\frac{1}{z_-} \nabla \bar{\mu}_- = \frac{1}{c_- z_-} [K_{0-} (\mathbf{v}_0 - \mathbf{v}_-) + K_{+-} (\mathbf{v}_+ - \mathbf{v}_-)] \quad (124)$$

We can apply Eqns(120 & 123) and derive the following equation of  $\nabla \bar{\mu}_-$  in terms of  $\nabla \mu_e$ .

$$\frac{1}{z_-} \nabla \bar{\mu}_- = -\frac{t_+^0}{v_+ z_+} \nabla \mu_e - \frac{F \mathbf{i}}{\kappa} \quad (125)$$

Where ...

$$\frac{1}{\kappa} = \frac{-K_{0-} t_-^0 - K_{+-}}{c_- z_- c_+ z_+ F^2} \quad (126)$$

By applying this expression for  $\nabla \bar{\mu}_-$  to Eqn(116) we derive the following expression.

$$-F \nabla \phi = \left( \frac{s_+}{nv_+} - \frac{s_0 c}{nc_0} + \frac{t_+^0}{v_+ z_+} \right) \nabla \mu_e + \frac{F \mathbf{i}}{\kappa} \quad (127)$$

This equation can be arranged to solve for  $\mathbf{i}$  as a function of  $\nabla \mu_e$ .

$$\mathbf{i} = -\kappa \nabla \phi - \frac{\kappa}{F} \left( \frac{s_+}{nv_+} - \frac{s_0 c}{nc_0} + \frac{t_+^0}{v_+ z_+} \right) \nabla \mu_e \quad (128)$$

As can be seen in the previous equation, we have been able to write an expression for the current density as a function of the gradient in the electrochemical potential. However, we would like to further reduce this expression such that it can be written explicitly in terms of concentrations.

Unlike other Equivalent Circuit Models of batteries (ECM), PDE based battery models must use physically meaning states, such as concentrations, to define such quantities as the state of charge (SOC) of a battery. To accomplish this, we need to express  $\nabla \mu_e$  as a function concentration. This begins by recognizing that we can express the electrolyte term as follows.

$$\nabla \mu_e = \frac{\partial \mu_e}{\partial \ln c} \nabla \ln c \quad (129)$$

Recall from  $\mu_e = \nu_+ \bar{\mu}_+ + \nu_- \bar{\mu}_-$  that electrochemical potential of the electrolyte can be written in terms of the positive and negative electrochemical potentials, each of which can be expressed via absolute activity, for  $i = \pm$ .

$$\begin{aligned} \mu_i &= RT \ln (c_i f_i a_i^\ominus) = RT \ln (\nu_i c f_i a_i^\ominus) \\ &= RT \ln (\nu_i) + RT \ln (c) + RT \ln (f_i) + RT \ln (a_i^\ominus) \end{aligned} \quad (130)$$

By taking the partial for each  $i = \pm$ , we arrive at the following two equations.

$$\frac{\partial \bar{\mu}_+}{\partial \ln c} = RT + RT \frac{\partial \ln f_+}{\partial \ln c} + z_+ F \frac{\partial \phi}{\partial \ln c} \quad (131)$$

$$\frac{\partial \bar{\mu}_-}{\partial \ln c} = RT + RT \frac{\partial \ln f_-}{\partial \ln c} + z_- F \frac{\partial \phi}{\partial \ln c} \quad (132)$$

By using Eqn(72), it is obvious that we can find the partial derivatives with respect to the  $\ln(c)$ , into which we can substitute the partial derivatives determined above.

$$\frac{\partial \mu_e}{\partial \ln c} = v_+ \frac{\partial \bar{\mu}_+}{\partial \ln c} + v_- \frac{\partial \bar{\mu}_-}{\partial \ln c} \quad (133)$$

After some involved substitution and simplification, we will arrive at the following expression.

$$\begin{aligned} \frac{\partial \mu_e}{\partial \ln c} &= vRT + RT \frac{\partial \ln f_\pm^\nu}{\partial \ln c} \\ &= vRT \left( 1 + \frac{\partial \ln f_\pm}{\partial \ln c} \right) \end{aligned} \quad (134)$$

This expression for  $\nabla \mu_e$ , can then be substituted into the Eqn(128), for current density.

$$\mathbf{i} = -\kappa \nabla \phi - \frac{\nu \kappa RT}{F} \left( \frac{s_+}{nv_+} - \frac{s_0 c}{nc_0} + \frac{t_+^0}{v_+ z_+} \right) \left( 1 + \frac{\partial \ln f_\pm}{\partial \ln c} \right) \nabla \ln c \quad (135)$$

We know from physics that conservation of charge in the electrolyte must hold. This can be expressed in gradient notation as ...

$$\nabla \cdot \mathbf{i}_e = 0 \quad (136)$$

We can substitute out expression for the current density of the electrolyte, into the conservation of charge, to yield the following equation. NOTE the addition of subscript  $e$  to denote that this expression pertains to the electrolyte.

$$\nabla \cdot \mathbf{i}_e = \nabla \cdot \left( -\kappa \nabla \phi_e - \frac{2\kappa RT}{F} \left( 1 + \frac{\partial \ln f_{\pm}}{\partial \ln c_e} \right) (t_+^0 - 1) \nabla \ln c_e \right) = 0 \quad (137)$$

In actual practice however, we assume that  $\partial f_{\pm} / \partial \ln c_e = 0$ , and additionally that we can lump all of constants together to obtain  $\kappa_D = 2\kappa RT (t_+^0 - 1) / F$ . Under these assumptions we derive the final equation which defines the conservation of charge in the electrolyte.

$$\nabla \cdot (-\kappa \nabla \phi_e - \kappa_D \nabla \ln c_e) = 0 \quad (138)$$

#### F. Butler-Volmer Equation

The final equation in the derivation of the microscale model which lays the foundation for the DFN model, is called the Butler-Volmer equation. This equation deals with the reaction kinetics of the battery and couples the four previous equations into a system of five equations. In principle, this equation defines rate at which lithium is transferred between the solid and electrolyte phases.

$$r = \frac{d \prod_i c_{\text{product},i}}{dt} \quad (139)$$

For linear dynamics, we can write the reaction rate as ...

$$r = k \prod_i c_{\text{reactant},i} \quad (140)$$

This constant is usually defined as a temperature dependent Arrhenius relationship such that we can write the reaction rate as follows.

$$r = k_f^0 c_{\text{reactant}} \exp \left( -\frac{E_a}{RT} \right) \quad (141)$$

Where  $E_a$  is the activation energy required to initiate the reaction.

1) *Activated Complex Theory*: In order for a chemical reaction to occur, there must be enough free energy to initiate and maintain the reaction. For the types of reactions seen in this paper, the process of going from reactants to products is only favored if the reactant possess a higher energy state than the final product of reaction. This positive delta in energy states maintains the chemical reaction, once it has been begun. However, there is often a barrier energy required to initiate such a reaction.

For the chemical reactions that occur in our lithium-ion battery, the reaction taking place is a reduction-oxidation (redox) reaction, which release/gains electrons. Unlike other redox reaction, the reaction in the battery is unique in that the objective of the reaction is to merely produce or consume electrons. However, in the presence of an electric field, the direction of the reaction can be preferentially determined by

the polarity of the given field across the cell.

Both the oxidation and the reduction directions of the reaction possess respective activation energies, which are required to drive the reaction into oxidation or reduction.

$$\Delta G_0 = -nF(\phi - \phi_0) \quad (142)$$

$$0 < \left| \Delta G_{\text{complex}}^\ddagger \right| < |\Delta G_0| \quad (143)$$

$$\begin{aligned} \Delta E_{a,o} &= E_{a,r} - E_{a,r}^0 - nF(\phi - \phi_0) \\ &= \Delta E_{a,r} - nF(\phi - \phi_0) \end{aligned} \quad (144)$$

$$\Delta E_{a,o} - \Delta E_{a,r} = -nF(\phi - \phi_0) = \Delta G_0 \quad (145)$$

To neatly capture this behavior, we define the asymmetric charge-trans coefficient  $0 < \alpha < 1$ , such that...

$$\alpha = \left| \frac{\Delta E_{a,r}}{\Delta G_0} \right|, \quad \text{and} \quad 1 - \alpha = \left| \frac{\Delta E_{a,o}}{\Delta G_0} \right| \quad (146)$$

By substituting the appropriate expressions into the equation for current flux, we arrive at the following reaction rates for both oxidation and reduction.

1. Rate of reduction:

$$\begin{aligned} r_r &= k_o^0 c_o \exp \left( -\frac{E_{a,r}}{RT} \right) \\ &= k_o^0 c_o \exp \left( -\frac{E_{a,r}^0 + \alpha nF(\phi - \phi_0)}{RT} \right). \end{aligned} \quad (147)$$

2. Rate of oxidation:

$$\begin{aligned} r_o &= k_r^0 c_r \exp \left( -\frac{E_{a,o}}{RT} \right) \\ &= k_r^0 c_r \exp \left( -\frac{E_{a,o}^0 - (1 - \alpha)nF(\phi - \phi_0)}{RT} \right) \end{aligned} \quad (148)$$

Since the reaction which is occurring must be charge neutral as the electrolyte is charge neutral, it is possible to show that the current flux for the oxidation and the reduction can be defined as a function of the corresponding reaction rates as  $i_r = -nFr_r$  and  $i_o = -nFr_o$ .

$$i_r = -nFk_o^0 c_o \exp \left( -\frac{E_{a,r}^0 + \alpha nF\phi}{RT} \right) \quad (149)$$

$$i_o = nFk_r^0 c_r \exp \left( -\frac{E_{a,o}^0 - (1 - \alpha)nF\phi}{RT} \right) \quad (150)$$

It should be noted that when the system is in equilibrium,  $\phi = \phi_{\text{rest}}$ . In order to capture how changes from the equilibrium potential effects the current flux, we define the **over potential** as  $\eta = \phi - \phi_{\text{rest}}$ . When properly substituted the previous equations, written in terms of the over potential are written as follows.

$$i_r = -nFk_o^0 c_o \exp\left(-\frac{E_{a,r}^0 + \alpha nF(\phi_{\text{rest}} + \eta)}{RT}\right) \quad (151)$$

$$i_o = nFk_r^0 c_r \exp\left(-\frac{E_{a,o}^0 - (1 - \alpha)nF(\phi_{\text{rest}} + \eta)}{RT}\right) \quad (152)$$

Before we continue manipulating these equations, note that when expressed via the overpotential  $\eta$ , the system is at equilibrium when  $\eta = 0$ .

Thus when in equilibrium, by using properties of the exponential function we can further simplify these oxidation and reduction currents to isolate the terms for applied potential and the activation energy.

$$i_r = -nFk_o^0 c_o \exp\left(-\frac{E_{a,r}^0}{RT}\right) \exp\left(-\frac{\alpha nF\phi_{\text{rest}}}{RT}\right) \quad (153)$$

$$i_o = nFk_r^0 c_r \exp\left(-\frac{E_{a,o}^0}{RT}\right) \exp\left(\frac{(1 - \alpha)nF\phi_{\text{rest}}}{RT}\right) \quad (154)$$

We know from the fact of charge neutrality of the system that sum total of the oxidation and reduction current fluxes must equate to zero.

$$i = i_o + i_r = 0 \quad (155)$$

Under the condition of equilibrium we can define the exchange current  $i_0$ , such that...

$$\begin{aligned} i_r &= -i_0 \exp\left(-\frac{\alpha nF\eta}{RT}\right) \\ i_o &= i_0 \exp\left(\frac{(1 - \alpha)nF\eta}{RT}\right) \end{aligned} \quad (156)$$

With this simplification, we can write express for the total ionic current  $[Am^{-2}]$  for both the oxidation and reduction directions of the reaction as...

$$i = i_o + i_r = i_0 \left\{ \exp\left(\frac{(1 - \alpha)nF\eta}{RT}\right) - \exp\left(-\frac{\alpha nF\eta}{RT}\right) \right\} \quad (157)$$

However, in battery literature, it is fairly uncommon to express the current flux in these units. Typically, we would express this quantity in molar form with unity  $[molm^{-2}s^{-1}]$ , by normalizing the current flux by Faraday's constant  $F$ .

$$j = \frac{i_0}{F} \left\{ \exp\left(\frac{(1 - \alpha)nF\eta}{RT}\right) - \exp\left(-\frac{\alpha nF\eta}{RT}\right) \right\} \quad (158)$$

2) *Exchange Current Density*: As defined in the previous section, the **exchange current density**  $i_0$  is defined under the condition of equilibrium. The physical interpretation of this term therefore, quantifies the amount of oxidation and reductions currents when the electrode is at rest. This term implies a type of dynamic equilibrium where in the electrons are being exchanged via the action of oxidation and reduction under no external simulation.

$$\begin{aligned} i_0 &= nFk_o^0 c_o \exp\left(-\frac{E_{a,r}^0}{RT}\right) \exp\left(-\frac{\alpha nF\phi_{\text{rest}}}{RT}\right) \\ &= nFk_r^0 c_r \exp\left(-\frac{E_{a,o}^0}{RT}\right) \exp\left(\frac{(1 - \alpha)nF\phi_{\text{rest}}}{RT}\right) \end{aligned} \quad (159)$$

Given this expression for both oxidation and reduction, we can define the following terms  $\mathcal{E}_{a,r}^0 = \exp\left(-\frac{E_{a,r}^0}{RT}\right)$  and  $\mathcal{E}_{a,o}^0 = \exp\left(-\frac{E_{a,o}^0}{RT}\right)$ . By equation  $i_0$  when defined using these symbols we get...

$$k_r^0 c_r \mathcal{E}_{a,o}^0 \exp\left(\frac{(1 - \alpha)nF\phi_{\text{rest}}}{RT}\right) = k_o^0 c_o \mathcal{E}_{a,r}^0 \exp\left(-\frac{\alpha nF\phi_{\text{rest}}}{RT}\right) \quad (160)$$

$$\frac{\exp\left(\frac{(1 - \alpha)nF\phi_{\text{rest}}}{RT}\right)}{\exp\left(-\frac{\alpha nF\phi_{\text{rest}}}{RT}\right)} = \frac{k_o^0 c_o \mathcal{E}_{a,r}^0}{k_r^0 c_r \mathcal{E}_{a,o}^0} \quad (161)$$

$$\frac{(1 - \alpha)nF\phi_{\text{rest}}}{RT} + \frac{\alpha nF\phi_{\text{rest}}}{RT} = \ln\left(\frac{k_o^0 \mathcal{E}_{a,r}^0}{k_r^0 \mathcal{E}_{a,o}^0}\right) + \ln\frac{c_o}{c_r} \quad (162)$$

$$\phi_{\text{rest}} = \frac{RT}{nF} \ln\left(\frac{k_o^0 \mathcal{E}_{a,r}^0}{k_r^0 \mathcal{E}_{a,o}^0}\right) \phi_\phi + \frac{RT}{nF} \ln\frac{c_o}{c_r} \quad (163)$$

$$\phi_{\text{rest}} = \phi^\ominus + \frac{RT}{nF} \ln\frac{c_o}{c_r} \quad (164)$$

Nerst Equation

$$\begin{aligned} i_0 &= nFk_o^0 c_o \mathcal{E}_{a,r}^0 \exp\left(-\frac{\alpha nF}{RT} \left(\phi^\ominus + \frac{RT}{nF} \ln\frac{c_o}{c_r}\right)\right) \\ &= nFk_o^0 c_o \mathcal{E}_{a,r}^0 \exp\left(-\frac{\alpha nF\phi^\ominus}{RT}\right) c_r^\alpha c_o^{-\alpha} \\ &= nFk_o^0 c_o^{1-\alpha} c_r^\alpha \mathcal{E}_{a,r}^0 \exp\left(-\frac{\alpha nF\phi^\ominus}{RT}\right) \end{aligned} \quad (165)$$

$$\begin{aligned} i_0 &= nFk_r^0 c_r \mathcal{E}_{a,o}^0 \exp\left(\frac{(1 - \alpha)nF}{RT} \left(\phi^\ominus + \frac{RT}{nF} \ln\frac{c_o}{c_r}\right)\right) \\ &= nFk_r^0 c_r \mathcal{E}_{a,o}^0 \exp\left(\frac{(1 - \alpha)nF\phi^\ominus}{RT}\right) c_o^{1-\alpha} c_r^{-\alpha-1} \end{aligned} \quad (166)$$

$$= nFk_r^0 c_o^{1-\alpha} c_r^\alpha \mathcal{E}_{a,o}^0 \exp\left(\frac{(1 - \alpha)nF\phi^\ominus}{RT}\right) \quad (167)$$

Since these two equations are equation, we can define an effective reaction rate constant.

$$k_0 = k_o^0 \mathcal{E}_{a,r}^0 \exp\left(-\frac{\alpha n F \phi^\ominus}{RT}\right) = k_r^0 \mathcal{E}_{a,o}^0 \exp\left(\frac{(1-\alpha)n F \phi^\ominus}{RT}\right) \quad (168)$$

Such that we can write the exchange current of the battery as

...

$$i_0 = n F k_0 c_o^{1-\alpha} c_r^\alpha \quad (169)$$

More nerst

$$\begin{aligned} k_0 &= k_o^0 \mathcal{E}_{a,r}^0 \exp\left(-\frac{\alpha n F \phi^\ominus}{RT}\right) \\ &= k_o^0 \exp\left(\frac{-E_{a,r}^0}{RT} - \frac{\alpha n F \phi^\ominus}{RT}\right) \\ &= k_o^0 \exp\left(\frac{-E_{a,r}^0}{RT} - \frac{\alpha n F}{RT} \left(\frac{RT}{n F} \left(\ln\left(\frac{k_o^0}{k_r^0}\right) + \ln \mathcal{E}_{a,r}^0 - \ln \mathcal{E}_{a,o}^0\right)\right)\right) \\ &= k_o^0 \exp\left(\frac{-E_{a,r}^0}{RT} - \alpha \left(\ln\left(\frac{k_o^0}{k_r^0}\right) - \frac{E_{a,r}^0}{RT} + \frac{E_{a,o}^0}{RT}\right)\right) \\ &= k_o^0 \left(\frac{k_o^0}{k_r^0}\right)^{-\alpha} \exp\left(-\frac{1}{RT} ((1-\alpha)E_{a,r}^0 + \alpha E_{a,o}^0)\right) \\ &= (k_o^0)^{1-\alpha} (k_r^0)^\alpha \exp\left(-\frac{1}{RT} ((1-\alpha)E_{a,r}^0 + \alpha E_{a,o}^0)\right). \end{aligned} \quad (170)$$

$$i_0 = n F k_0 \left(\prod_i c_{o,i}\right)^{1-\alpha} \left(\prod_i c_{r,i}\right)^\alpha \quad (171)$$

$$\prod_i c_{o,i} = c_e (c_{s,\max} - c_{s,e}) \quad (172)$$

Now that we have formulated this quantity in terms that make sense intuitive sense for the system, we can now rewrite these equations in terms that make sense in the context of battery specific applications. Namely, we can rewrite the expression for overpotential  $\eta$ , in terms of the batteries open circuit potential, and the potentials applied to the solid and electrolyte phases, where  $(U)_{ocp} = \phi_{rest}$

$$\eta = (\phi_s - \phi_e) - U_{ocp} \quad (173)$$

Written in both SI, and molar notion the final Butler-Volmer equation is...

$$i = F k_0 c_e^{1-\alpha} (c_{s,\max} - c_{s,e})^{1-\alpha} c_{s,e}^\alpha \left\{ \exp\left(\frac{(1-\alpha)F}{RT} \eta\right) - \exp\left(-\frac{\alpha F}{RT} \eta\right) \right\} \quad (174)$$

$$j = k_0 c_e^{1-\alpha} (c_{s,\max} - c_{s,e})^{1-\alpha} c_{s,e}^\alpha \left\{ \exp\left(\frac{(1-\alpha)F}{RT} \eta\right) - \exp\left(-\frac{\alpha F}{RT} \eta\right) \right\} \quad (175)$$

$$\begin{aligned} i_0 &= F k_0 c_e^{1-\alpha} (c_{s,\max} - c_{s,e})^{1-\alpha} c_{s,e}^\alpha \\ &= F \underbrace{k_0 c_{e,0}^{1-\alpha}}_{k_0^{\text{norm}}} \left(\frac{c_e}{c_{e,0}}\right)^{1-\alpha} \left(\frac{c_{s,\max} - c_{s,e}}{c_{s,\max}}\right)^{1-\alpha} \left(\frac{c_{s,e}}{c_{s,\max}}\right)^\alpha \end{aligned} \quad (176)$$

$$j = k_0^{\text{norm}} \left(\left(\frac{c_e}{c_{e,0}}\right) \left(\frac{c_{s,\max} - c_{s,e}}{c_{s,\max}}\right)\right)^{1-\alpha} \left(\frac{c_{s,e}}{c_{s,\max}}\right)^\alpha \times \left\{ \exp\left(\frac{(1-\alpha)F}{RT} \eta\right) - \exp\left(-\frac{\alpha F}{RT} \eta\right) \right\} \quad (177)$$

#### IV. SINGLE PARTICLE MODEL (SPM & SPME)

As mentioned previously the microscale and macroscale models described above are computationally very expensive to compute numerical solutions for. Due to this property, these models are often not used as a practical work-horse models for the daily needs of systems that incorporate battery models, and that model like the DFN formulation, are generally used as benchmarks of accuracy and performance for a reduced-order or BLANK battery models. These simplified models usually sacrifice a great deal of the accuracy which the DFN model worked so hard to preserve from first principles. As a consequence, model such as a simple lookup-table or equivalent circuit model (ECM) are dominately purposed around the use of real-time systems that need a roughly accurate computation about the state of charge of a cell. However, there are use-cases where these simplified models are too low resolution, and require higher fidelity models, that are slightly more expensive to compute, but provide much better tracking to the DFN base-line model.

Two such model variations are the Single Particle Model (SPM) and the Single Particle Model with Electrolyte Dynamics (SPMe). These models are far an away the work-horse of many battery control and estimation tasks. The guiding principle of these models is to focus on the dominate dynamics of the battery. It is a well known fact presented in undergraduate control theory that the slowest pole/zero will dominate the transient dynamics for a given linear system. For lithium ion batteries, it can be shown that the diffusion of lithium into the solid-phase (electrode) evolves ths slowest, and therefore dominates the dynamic behavior of the battery for the time-scales which we are mostly interested in. The SPM model takes this fact and reduces each electrode, of the battery, to an individual sphere of active material that is presumed to be representative of the average dynamics of the electrodes. It should be notes that in this model, another simplification is made whereby the electrolyte concentration and dynamics are discarded. The SPMe elevates the SPM model by including these electrolyte dynamics back into the model, allowing this model to capture the nonlinearities which are present due to transport behavior of the electrolyte.

#### V. SPM MODEL DERIVATION

The previous system of governing partial differential equations, was derived using the microscale interactions and mechanisms. While this approach to analysis is particularly informative about the how these sorts of high fidelity models are derived, solving them for an entire cell (using a system of thousands or millions of coupled PDEs) is computationally infeasible, even for modern computers. There are many different approaches to creating macroscale models that capture the first principle mechanisms of action,

but are far more computationally viable. This simplification comes with a penalty on the accuracy of model; however, these models are still far and away some of the most high fidelity battery models which are routinely used today to check

In particular, the Doyle-Fuller-Newman (DFN) model uses a technique of volume averaging to be able to solve governing equations which were derived above, over larger domains, approximating (on average) the behaviors of each electrode and the electrolyte.

## XII. BIOGRAPHY SECTION

## VI. ANALYSIS OF GOVERNING EQUATIONS

## VII. NUMERICAL SIMULATION

## VIII. RESULTS

## IX. CONCLUSION

The conclusion goes here.

## ACKNOWLEDGMENTS

This should be a simple paragraph before the References to thank those individuals and institutions who have supported your work on this article.

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