

# Calcite Palaeopiezometry

## Laboratory notes

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*Supervised by Ernest Rutter and Steve Covey-Crump*

**This document is composed of two parts:**

Part 1: Progress log – Organisational notes, progress updates, planning.

Part 2: Handwritten laboratory notes – complete record of thought development and workings.

All entries are dated, and no entries were edited or removed.

# Calcite Palaeopiezometry

## Progress log

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## Instructions:

Date all the entries, do not edit or delete past entries. When adding comments, date them. Name the relevant files and their locations.

## 29<sup>th</sup> October 2020

*Edit by Kamil on 30/10/2020:*

Only just started looking into the project. I had a meeting with Ernest sometime in June, but I forgot most of what was said. I remember struggling with understanding what the project is about exactly. Ernie sent me some papers to read, but I found them somewhat challenging. I have all the data.

I have decided to:

- Learn a bit more about crystallography. Firstly, by reading the Earth Materials textbook by Klein and Philpotts.
- Learn a bit more about calcite
- Read papers listed by Ernie
- Find more relevant literature

So far, I have read chapter 4 of Klein and Philpotts, and understand (I think) the Miller indices. This is a bit more complex for the 4-axis hexagonal system that calcite is a part of.

I have started a ‘lab-book’ – just a notebook with all the raw notes I collect.

## 30<sup>th</sup> October 2020

*Edit by Kamil on 30/10/2020:*

Continued reading of the Klein and Philpotts textbook – I learned about the 4-axis hexagonal Miller indices, and about how the stereonet projection works for hexagonal crystal systems. Identified calcite as  $\bar{3} 2/m$  in the Hermann-Mauguin notation.

Learned a bit about how stereonets for hexagonal systems work. Saw examples of rhombohedron and hexagonal scalenohedron - though I’m not sure which ‘form’ I should care about for the analysis – or perhaps there is more? The symmetry is the same, but stereonets look different (face projections).

## **31<sup>st</sup> October 2020**

*Edit by Kamil on 31/10/2020:*

Understood what chirality means for scalenohedron form, and how stereoplots are projected from upper hemisphere – also how crystal axes always go through the edges of the crystal.

Got myself a few rhombohedral calcite blocks to help with visualisation.

## **3<sup>rd</sup> December 2020**

*Edit by Kamil on 05/12/2020:*

I had a zoom meeting with Steve. I explained where I am standing, and explained how the more I read, the less I feel know. We spoke about some of the experiment details, and about some of the geometry. Steve helped me paint the bigger picture.

Steve suggested that work & reading should go in pair, and that I should try to analyse the data as I get to it – without necessarily fully understanding the next step or having a complete image of the bigger picture. I think he might be right.

Steve suggested I set up a zoom meeting with Ernie. I am trying to get that arranged for sometime in the next 2 weeks.

## **22<sup>nd</sup> December 2020**

*Edit by Kamil on 04/03/2021:*

Had a zoom meeting with Ernie. We discussed (fill)

## **4<sup>th</sup> March 2021**

*Edit by Kamil on 04/03/2021:*

Last week I submitted an assignment for a different module which focused on EBSD, a technique used to tell the orientation of grains. I feel like I understand it well now.

Today I re-watched Ernie's first lecture and made some better notes on orientation presentation, Turner method, and yield stress criteria/analysis.

## **5<sup>th</sup> March 2021**

*Edit by Kamil on 08/03/2021:*

I re-watched Ernie's second lecture and made more notes on topics touched.

## **8<sup>th</sup> March 2021**

*Edit by Kamil on 08/03/2021:*

I made some detailed notes on EBSD – however there is an even better account of the EBSD technique in the piece of work I wrote for Analytical Techniques 2 weeks ago.

## **9<sup>th</sup> - 15<sup>th</sup> March 2021**

*Edit by Kamil on 15/03/2021:*

Over the past days I have re-read the relevant literature and took notes. I feel like I know how to tackle the data, however I am still not 100% clear on the exact parameters used during the experiment. I need to work on that next.

## **23<sup>rd</sup> March 2021**

*Edit by Kamil on 23/03/2021:*

Got an email from Ernie outlining the 'disciplinary' plan for getting the project done. To be honest this is quite useful, as I am lost in the maths and the order things come in. He has provided solutions for quite a few of the grains – this may be good framework for what I'm meant to do. And then to somehow link it to the stress. I need to read his description on a soon basis.

We agreed that I will produce preliminary background + intro sections for April 6<sup>th</sup>.

## **24<sup>th</sup> March 2021**

*Edit by Kamil on 24/03/2021:*

Yesterday I asked Ernie some of the questions I had written down. They're regarding stress homogeneity, finding e+a/h fabric in natural rocks, and recrystallisation. Ernie provided answers, with reference to the Spiers 1980 paper. Steve 'translated' the bit on stress homogeneity and how I can apply it to my project – either I observe it and it's relevant, or I don't and it's part of background.

## **26<sup>th</sup> March 2021**

*Edit by Kamil on 26/03/2021:*

I read the document on stress and strain. I was hoping to also read the Spiers paper and make notes on both, but ran out of time.

## **27<sup>th</sup> March 2021**

*Edit by Kamil on 28/03/2021:*

I covered 20 pages of the Pet Engineering document, and still need to look at a few sections of (Intracrystalline plasticity) etc.

## **5<sup>th</sup> April 2021**

*Edit by Kamil on 05/04/2021:*

Took a few day long break for Easter and to complete other coursework. Yesterday I finished the engineering document, and today I had a look at the detailed work-through of the Calcite data that Ernie sent me as aid. I also read the Spiers 1980 paper which addresses the stress homogeneity and strain heterogeneity in deformed samples – and Ernie actually touched on how cross-sectional estimation of stress may be a bit too simplistic. I'll need to look at it later.

I looked into running MTEX on my Matlab, but with no result. I am getting an error when trying to input data. I updated my Matlab but still get the same error.

I think I will have to do without MTEX – perhaps with just other stereonet plotting software and drawing package touch-ups like Ernie suggested.

## **7<sup>th</sup> April 2021**

*Edit by Kamil on 07/04/2021:*

Yesterday I worked on the Euler angle problems – still figuring out which way is up.

Today I finally created the dissertation word file. I made an outline of what goes in each section, but this will need to be updated with details from correspondence with Ernie – I've already noticed it answers some questions I had, but won't have the time to do it today.

## **11<sup>th</sup> April 2021**

*Edit by Kamil on 12/04/2021:*

I have looked into the spherical approximation for stress that Ernie spoke about – it is simple enough to apply mathematically. I was initially worried that I would have to struggle with grains located off-axis, but Ernie re-assured me that because of the geometry and scale of the section, it is okay to assume all grains are effectively on-axis.

## **12<sup>th</sup> April 2021**

*Edit by Kamil on 12/04/2021:*

Today I looked into the Bunge convention and all the files provided by Dave Wallis. I noticed that we can't seem to agree on what Bunge really means – I think all angles should be right-handed (anticlockwise), but the manual for the CHANNEL software defines E1 and E3 as opposite sense to E2. Some other source I found on the internet lists them all as right-handed. I asked Ernie for input.

## 13<sup>th</sup> April 2021

*Edit by Kamil on 13/04/2021:*

In the morning I figured out the Bunge conventions. Does not actually matter how I take them, as sample is axi-symmetric. In the worst case scenario, I'll have a reflection of what Ernie worked on, but both are equally valid. In fact, the stereonet plot I produced is rotated by 90 degrees anticlockwise compared to Ernie's.

In the evening I figured out the spherical cap correction for stress trajectories/isobars. It was actually quite a pleasant day of coding.

## 14<sup>th</sup> April 2021

*Edit by Kamil on 14/04/2021:*

I extracted the stress profile of my sample, and matched the grains to it, so now I know the approximate stress that each grain experienced. I also made some diagrams which may be useful for the dissertation itself.

I noticed that the assumption about strain being largely irrelevant for sample geometry may not be strictly true, as even though the strain is  $\sim 1\%$ , it's not homogenous. The central part is more strained. I did ask Ernie if I should account for this, but apparently this sort of correction is minor compared to uncertainties introduced by stress heterogeneities due to the grains themselves.

I also had a read of the document guidelines. I need to adjust my font and font size, but also look at the 'figure list' section.

## 15<sup>th</sup> April 2021

*Edit by Kamil on 15/04/2021:*

I started preparing the channelling-montage and the band-contrast images for the twin trace extraction exercise – I made both images into one long record and I labelled the grains. Some are problematic, as I discovered the two images do not exactly match. I asked Ernie why. I think it is to do with the geometry at which they were collected – but that still does not explain why some small grains are present on one image and not the other and vice versa.

I wanted to index the images with sample dimensions in matlab today, but my computer decided to force a loooong windows update, so that's a task for tomorrow.

## 20<sup>th</sup> April 2021

*Edit by Kamil on 20/04/2021:*

Today I plotted the twin c-axes onto the stereonets and found a way to connect them with arcs. I verified the host c-axis to twin c-axis angle, which should be  $52^\circ$ , and in most cases is. Grain 26 shows an angle of  $83.85^\circ$ , which like Ernie suggested, is probably an anomaly in the measurement.

The goal for tomorrow is to figure out how to get the rotation towards the sigma 1. I'm thinking of doing a rotation matrix, but I think it will be a bit of trial and error.

## 21<sup>st</sup> April 2021

*Edit by Kamil on 21/04/2021:*

I found something called “Rodrigues’ rotation formula”. It allows for easy rotation of a unit vector around a specified axis. Very useful for going to the e-plane by going from 26 degrees from c-axis first (towards m-axis), and then going 60 degrees around the original c-axis so that I index with the c-e plane mirror axis, not the r-plane containing part of it.

I wrote two matlab functions – one that extracts cosines from Euler, and another that carries out the Rodrigues’ rotations. I tested them on a few data points to see if I can get from host orientation alone to the measured twin. I can. One of the three solutions will be within  $2^\circ$  and others are about  $85^\circ$  away. I find that the solution tends to be the one at  $240^\circ$  index, except when two twins are present – then it is common that solution is at  $180^\circ$ , not  $2^\circ$ , away – this is a symmetry consideration for later.

**22<sup>nd</sup> April 2021**

*Edit by Kamil on 22/04/2021:*

In the morning I confirmed that once the hosts are rotated so that c-axis is coincident with the twin's (as measured by EBSD), the remaining axes can also be aligned by either 0, 1, or 2 rotations by 120°.

I wrote some code that travels a specified number of degrees from the c-axis towards the c-e plane, and then rotates 120 degrees twice, giving me orientations of all possible twin c-axes (but not all twin axes, as there are still three possibilities for each of 3 axes – this isn't exactly relevant because of symmetry though). Aim is to use this later to simulate the three possibilities and choose the best one based on the image – i.e. align twin traces. I will also need to do the same for the r-planes, which should be an easy adaptation of the same code.

Next, I wrote a piece of code that allows me to translate any vector expressed as direction cosines into plunge/azimuth format, so I can plot it on my stereonets. I checked the results against plotting directly from Euler, and method seems to work fine.

Because I wrote so many matlab functions by now, I made a word file which summarises the library of code I created – this will aid me in finding the right function when I need it.

*Edit by Kamil on 22/04/2021, late evening:*

I worked out how to index and find the right solution for the EBSD measured twins – and how I can then go 72 degrees (26+45) from c-axis in that direction along c-e plane to find the sigma 1 direction – I plotted those on the stereonet. They make sense. Grain 26 is removed from analysis, and so may be grain 135 – it indexes 10 degrees away from measured twin. Most other grains do so within 2 degrees. The 10 degrees are quite noticeable on the stereonet.

## 23<sup>rd</sup> April 2021

*Edit by Kamil on 23/04/2021:*

I worked out sigma 3. Then I plotted it on the stereonets and did contours. It shows roughly what I expected. Spread out around a great circle perpendicular to sigma 1, albeit also slightly tilted. Ernie thinks this tilt of  $\sim 20^\circ$  may be significant and suggested it's due to pre-existing fabric. I think it may be because the sample was mis-aligned on the EBSD stage. It's something I need to keep in mind.

While sigma 3 is at  $-19^\circ$  to c-axis, I used the value of  $-18.9^\circ$  to avoid ambiguity with plotting arcs between sigma 1 and sigma 3 on the stereonet – as I always choose the acute angle.

I also tidied up the textfile tables – I wrote a piece of code that helps with their formatting, so they should be more readable.

I also devised a method for getting the potential trace of the e-plane out of the host orientation, but I have not yet had the time to implement it.

## 27<sup>th</sup> April 2021

*Edit by Kamil on 27/04/2021:*

I plotted the second twin sigma 1, sigma 3, and twin c-axis. It works fine, but I need to think about the implications of the  $109.94^\circ$  separation between the two predicted sigma 1 directions. This is expected – but what I need to consider for the Schmid factor analysis later, is that on top of the angle in the c-e plane, I need to consider the angle that the sigma 1 vector makes with the c-e plane, as that's where the slip vector is – something to think about.

I used the Miller-Bravais indices to confirm the angular relationships in calcite. That is, the  $26^\circ$  between c-axis and e-pole,  $44^\circ$  between c-axis and r-pole, and  $63^\circ$  between the c-axis and f-pole.

## 28<sup>th</sup> April 2021

*Edit by Kamil on 28/04/2021:*

I worked out the e-poles, r-poles, and their corresponding traces on the image plane. Doing the cross product of e-pole with  $[0,0,1]$  – the image z-axis, did work fine to produce traces.

I tested this with Euler angles of zero – results are as expected. I then tested it against a few grains with known twin orientations. Both methods produce the same solution index, so I will be able to find the correct of the three solutions for each twinned grain.

I made some corrections – previously images were squished, sometimes showing small angle discrepancies. Now this should not happen.

I changed the approach to collecting traces as well – instead of plotting the trace figures on top of the image using grain coordinates, I instead generated ‘expected traces’ figure for each grain, and will overlay this on top of the channelling image in Illustrator – this approach is more manual, but gives me more freedom in moving the axes around the image, which will allow me to better judge the angles.

After plotting a few grains and seeing planes that I don’t expect to see, I decided that I will include the f-slip system as well – just to see if it helps to determine what is and what is not an e-plane.

## 29<sup>th</sup> April 2021

*Edit by Kamil on 29/04/2021:*

Went through the first 164 grains, with 159 left for tomorrow (hopefully). I made a list of all the grains that have problems with them. So far, the ‘problem’ list includes 54 grains (33% of all analysed). Some of the problems are minor, such as small angle discrepancies, or inability to see small grains properly (they likely are untwinned). Bigger problem is the ambiguity when one or two of the potential solutions align with themselves, or with one of the f or r planes.

I found that e-planes from smaller grains seem to propagate through bigger grains. Sometimes even through entire image. Perhaps this is because the energy required to twin a small grain is enough to fracture the surrounding grains too? Or perhaps those lines are just artifacts.

## 30<sup>th</sup> April 2021

*Edit by Kamil on 30/04/2021:*

Gathered twin solutions for the remaining grains. 108 of the 323 are ‘problematic’. Roughly 1/3<sup>rd</sup>. That’s still quite an improvement. In total there is

341 twin planes now, but this includes the problematic ones too. I will have the number of good twins soon.

## **1<sup>st</sup> May 2021**

*Edit by Kamil on 01/05/2021:*

Generated all the twins and picked out only the solutions that I observed. Then I developed code that translates direction cosines back into Euler angles, as this will be easier to feed into the code that already exists. I double checked it all translates back into the cosines the way I want to – it does.

I sent Ernie all the twinned grains I gathered – 216 of which are non-problematic.

## **2<sup>nd</sup> May 2021**

*Edit by Kamil on 02/05/2021:*

Wrote code that removes the ‘problematic’ grains identified earlier. This needs to be fed a list of problematic grain numbers, which makes it less universal for future datasets but makes it more robust for different table arrangements – in fact all it needs is for Grain No. to be in columns 1.

I completed the Turner-Weiss analysis – on the 215\* new grains with 247 new twins, and hence 247 sigma 1 and sigma 3 predictions. Data is good. I seem to have picked up more sigma 1 sticking out of the image than ‘up’ or ‘down’ on the image – although these should be symmetric.

I think it’s a sampling bias, but not 100% sure. May also be that deformation was not perfectly axi-symmetric, or could be a pre-existing fabric in the rock.

\*Seems I missed one grain on the “Problematic grains” list, that I flagged down as problematic in the excel file.

## **3<sup>rd</sup> May 2021**

*Edit by Kamil on 04/05/2021:*

I plotted all the e-poles, as well as only the resolved ones to confirm that what I saw before is just a sampling bias – it probably is.

## 5<sup>th</sup> May 2021

*Edit by Kamil on 05/05/2021:*

I started the day by figuring out how to do the inverse pole figures – the method works, but I think data needs to be adjusted for Schmid factor – else it plots all over the place.

Next, I tried working out the Schmid factor. Initial working has given me 64 grains with positive Schmid factor. That is low. I plotted them, and it happens to be the ones that should not be twinned. Perhaps it is the negative Schmid factor that I am looking for. I am trying to work it out, but the task will probably extend to tomorrow.

Solved it – I am in fact looking for negative Schmid factors in the sense that it is calculated along acute angles, but my slip vector will always be obtuse. Hence doing  $180^\circ - \lambda$  solves it, but has the same effect as switching the Schmid factor signs.

## 6<sup>th</sup> May 2021

*Edit by Kamil on 07/05/2021:*

Yesterday I counted all the twin planes in each valid grain. Easy, but time consuming.

## 7<sup>th</sup> May 2021

*Edit by Kamil on 08/05/2021*

Yesterday I fixed the two problematic grains: 181 and 283. I exported grain sizes, twin counts, and pressures to a spreadsheet.

## 8<sup>th</sup> May 2021

*Edit by Kamil on 08/05/2021*

I worked out the apparent and corrected twin densities – by using the feret diameter. Then to adjust for the dip of the twin planes relative to the plane of the image, I divided the apparent densities by the cosine of the plunge the e-pole makes with the plane. I'm now ready to split it into groups and plot.

## 10<sup>th</sup> May 2021

*Edit by Kamil on 11/05/2021*

I plotted various graphs showing relationships between the Schmid Factor, Pressure, Twinning activity, and grain size. I found that even though there is now plenty more data available, the relationships are not so clear cut. The best I found is a data-bounding envelope on the activity-pressure charts. This may or may not depend on the Schmid factor for the larger grains.

## 11-21<sup>st</sup> May 2021

*Edit by Kamil on 23/05/2021*

This time has been spent producing figures and writing the thesis itself.

## 27<sup>th</sup> May 2021

*Edit by Kamil on 27/05/2021*

Poster and presentation complete. Paper notes scanned in. All files available at:

<https://github.com/Kamkosiorek/Calcite-Palaeopiezometry>

<https://drive.google.com/drive/folders/1Bh9hmtb2ONqP3WhJnVcO5yk0Qjknu6pj?usp=sharing>

# Earth Materials Chapter 4

X-ray diffraction  $\neq$  x-ray powder diffraction

perfected crystal ~~imperfect~~,  
atomic distances learned.

Atomic radius:  $\frac{1}{2}$  distance between 2 identical atoms (centres, spherical)

Coordination number = number of anions

$\text{Ca}^{2+}$  often forms octahedral to cubic coordination  
 $\text{e}_\text{VI}$        $\text{e}_\text{VIII}$

Isodemic - All bonds are the same strength

Calcite  $\rightarrow$  Anisodemic?

$\text{CaCO}_3$        $\text{e} (\text{CO}_3)^{2-}$        $\Delta$  - triangles  
polygons more strongly bonded.  
to each other than other ions

Figure 4.9

Calcite structure: Combination of Octahedral  $\text{CaO}_6$   $\frac{2}{3}$  charge?  
and Triangular  $(\text{CO}_3)^{2-}$   $\frac{1}{3}$  per O  
 $\frac{-2/3}{\text{per O}}$

Each oxygen coordinated to Calcium AND Carbon

Some mention of ionic substitution but I think  
it is not relevant.

30 OCT 2020

1 1 1 .  
-1 1 1 .  
1 -1 1 .  
1 1 -1 .  
-1 -1 -1 .

$\{001\} \rightarrow (001), (00\bar{1})$

But still true for hexagonal?

# Earth Materials Chapter 5

Hermann - Nagrin

Calcite: Trigonal,  $\bar{3}m$ , H-N:  $\boxed{\bar{3} \ 2/m}$  Space grp:  $R\bar{3}c$

4 types of symmetry in - Mirror plane,  $n$  = number of repeats

$R$  - Rotation axis

$i$  - Centre of symmetry  $\rightarrow$  Inversion

$\bar{R}$  - Rotoinversion axis  $\rightarrow$  rotation with inversion

$C$  completed

Point group elements  $\rightarrow$  symmetry elements intersect in the centre of the crystal.

32 possible crystal classes

$\bar{3}$  equivalent to 3-fold rotation in combination with centre of symmetry (:- inversion)  
 ↗ rotational axis perpendicular to mirror

$1\bar{R}_3, 3R_3, 3m$

$\bar{3} 2/m$  is Hexagonal  $1\bar{R}_3 = 1R_3 + i$

? Rotate 3 times  $\rightarrow$  like turbine, and invert looking down?

In 5 of 6 systems, 3 axes are used to describe crystal

In 1 of 6, Hexagonal, 4 axes are used:

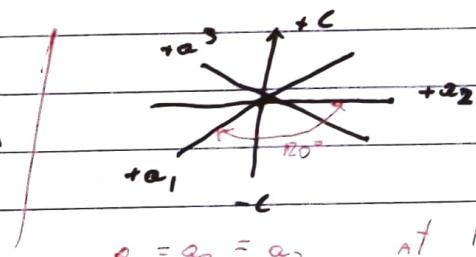
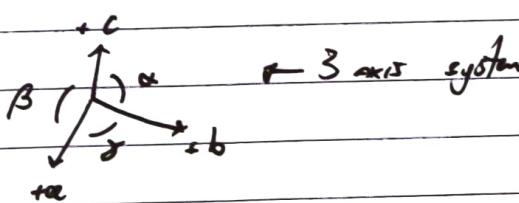
$a_1, a_2, a_3, c$

$hkl$

for vertical  
intercept,  $g$ onal  
direction

$(h\bar{k}\bar{l})$

for hexagonal



$a_1 = a_2 = a_3$  at  $120^\circ$

$c$  perpendicular to plane

Miller indices  $\rightarrow$  inverse of Weiss system.

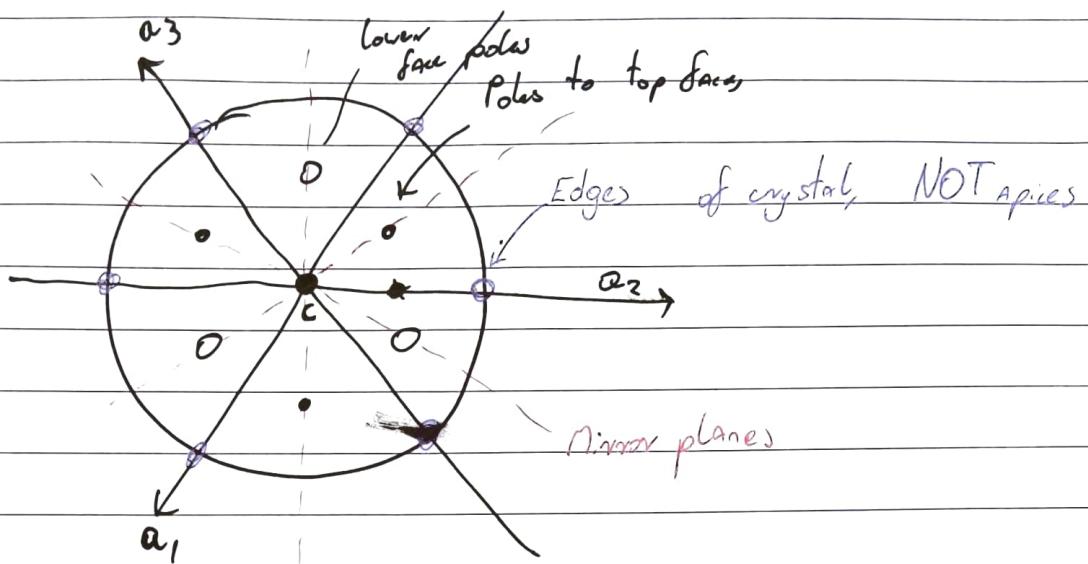
Just coordinates of surface  $(\infty, \infty, 1) \rightarrow (0, 0, \bar{1})$

Diffn for Hexagonal sysy



$\bar{3}2/m \rightarrow$  rhombohedron.

Parallel faces by inversion, not mirror reflection



Clinographic projections  $\rightarrow$  perspective drawings of crystals,  
much better than photograph

- c-Axis vertical, b axis East - west, a Axis towards observer

1.  $\bar{3}2/m \leftarrow$  c axis is rotoinversion, and thus horizontal axes are 2-fold ~~not~~ rotation.

General form  $\{h\bar{k}\bar{l}\}$  is a scalenohedron

$\hookrightarrow$  Closed form of 12 scalene triangles

Distinguished from ~~dipyramid~~ by zig-zagging middle edges

Common form, rhombohedron  $\rightarrow \{h0\bar{l}\}$  and  $\{0\bar{h}\bar{l}\}$

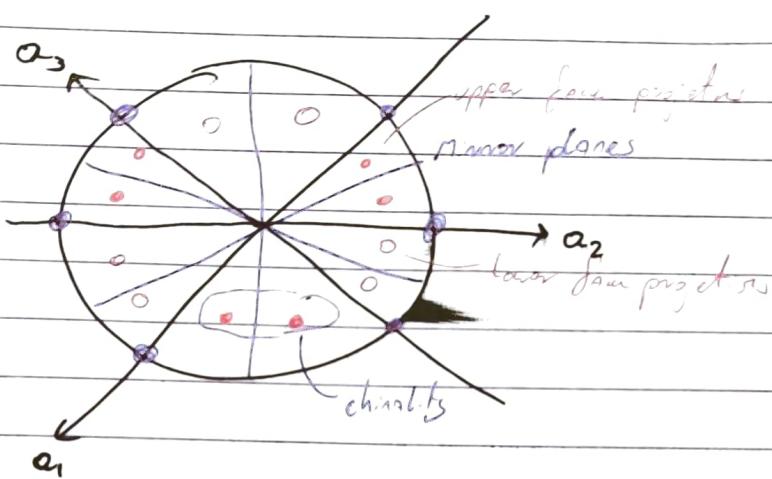
Closed form of 6 rhomb shaped faces

- like a squished or extended cube.

~~Rhombohedron~~ common index =  $\{10\bar{1}1\}$

In general form: (Hexagonal Scalenohedron  $\rightarrow$  6-edge

di-pyramid  
with zig-zag edges



XRC - X-ray crystallography

Different forms of twining depending on  $\rightarrow$  fibrous, granular, lamellar or 'compact'

Clearance in 3 directions parallel to rhombohedral form

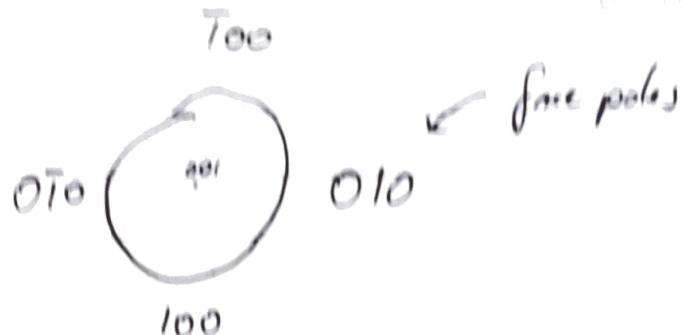
Scalenohedral faces are chiral (mirror image symmetry).  
Rhombohedral faces are achiral.

Calcite  
w.k. page

## In crystallography



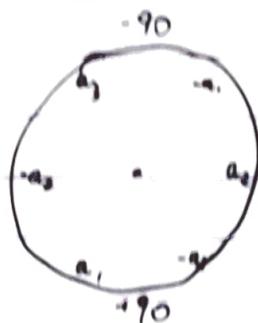
Projected from upper hemisphere onto horizontal circle,  
(p + m, hkl, d, l)



But once again for 3 axis system

Wulff nets more commonly used. in crystallography

Idea from the web:



- Solid bold line = m axis
- c axis is perpendicular to page

No horizontal (printing arch)  
for  $\bar{3}2/m$

Vertical axis  $\rightarrow$  ~~one~~ 3-fold rotation-inversion axes. no perpendicular  
mirror axis vertical and intersect with horizontal axis?

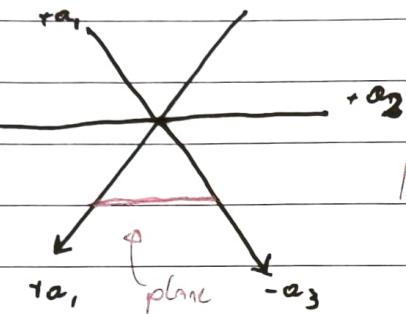
Figure 5.9A + 5.9B

In Hexagonal system  $\rightarrow$  Use Miller-Bravais <sup>index</sup> notation



Pinnacoid - Hexagonal prism  
surfaces on top and bottom

Plan view



Intercepts at  $(1, 0, 0), (0, 1, 0), (-1, 0, 1)$

$\hookrightarrow (1, 0, 0, 1)$

Simple enough

Also  $\{hk\bar{l}\}$  for general form  $\leftarrow$  but still not sure  
HOW that works.

Wiki: Notes: Calculate it: Dihedral scaleno hexagonal. Centrosymmetric  
Space group number: 162-167

What is form index?

$[001]$  <sub>→</sub> direction,  $(001)$  faces

perpendicular to

$\{001\}$  form

↳ can mean any combination of  
positive and negatives?

Creation of stereographic projection of a crystal first  
involves physical projection of perpendiculars to each of the crystal faces  
(face poles)

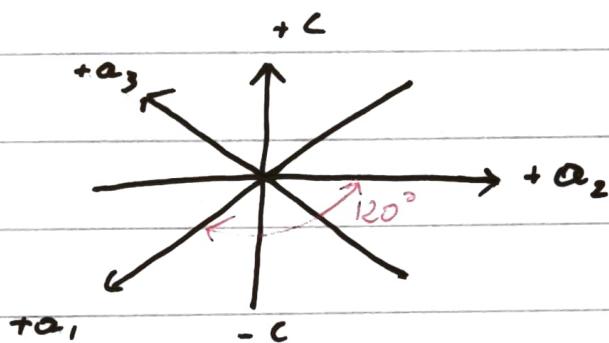
$\{001\} \rightarrow (001), (00\bar{1})$

$\bar{3}n$  is abbreviation of  $\bar{3}2/m$

Notes on calcite & crystal structures  
in general from yesterday:

Calcite is in hexagonal crystal structure

- Has 4 crystal axes - c, and  $a_1$ ,  $a_2$ ,  $a_3$



$a_1$ ,  $a_2$ ,  $a_3$  are co-planar, c Axis is normal

Calcite is anisodemic (?) - not all bonds are the same strength. Each oxygen is coordinated to Calcium AND Carbon

Hermann-Mauguin notation:  $\bar{3}$   $2/m$   $\text{C}$   
c H-M

2-fold rotational axis  
perpendicular to mirror axis

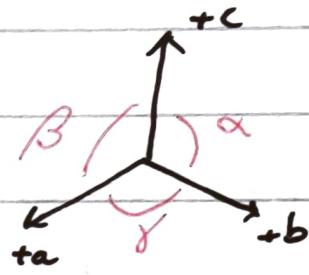
$\text{C}3$ -rto inversion axis

Space group:  $R\bar{3}c$

So - Rotate over 3 times and invert looking down?

30 OCT 2020

For system with 3 axes (NOT Calcite):



indicates negative axis

Weiss

Miller

And Miller indices are:  $(\infty, \infty, 1) \rightarrow (0, 0, 1)$

$\begin{matrix} a & b & c \end{matrix} \rightarrow \begin{matrix} 0 & 0 & 1 \end{matrix}$

Symmetry Key: R - Rotation axis

m - Mirror axis

i - centre of symmetry, inversion

$\bar{R}$  - Rotoinversion - rotation with inversion

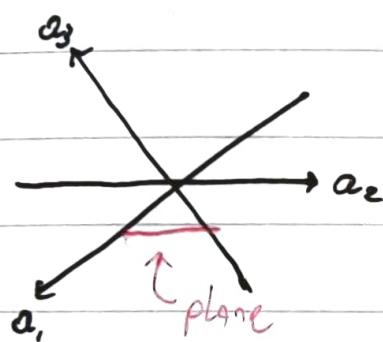
Calcite is also  $\bar{3}m$  - I think simpler, but less complete description than H-M

(Earth Materials, Chapter 4 & 5)

## Notes on calcite and stereonet projections

In crystallography, upper Hemisphere, Wulffnet projections comes.

In hexagonal systems, use Miller-Bravais index:



Intercepts at  $(1, \infty, -1, \infty)$   
 $\hookrightarrow (1, 0, \bar{1}, 0)$   
 in Miller-Bravais format

$(001)$  is a face

$[001]$  is a direction normal to that face - "face poles"

$\{001\}$  is a 'form'. Mean possible +/- combinations:  $(001)$ ,  $(00\bar{1})$

In a basic stereographic projection, face poles are projected.

$\bar{3}m$  is just abbreviation of H-M's  $\bar{3}2/m$

Clinographic projections - perspective drawing of a crystal.

C-Axis vertical, b-Axis East-West, a-Axis towards observer.

(Earth materials, Chapter 4 & 5)

31 OCT 2020

## Few notes on calcite crystal structure/shape

- Few shapes can satisfy the  $\bar{3}2/m$  symmetry.
- Common ones are rhombohedron and hexagonal scalenohedron,  
(or combination of the two) (6-faced edge dipyramid  
with 2:3-2:4:3 middle edges)
- Cleavage in 3 directions parallel to rhombohedral form.
- Scalenohedral faces are chiral (mirror image symmetry)
- Rhombohedral faces are achiral
- Different forms of twinning common in calcite:  
Fibrous, granular, lamellar, compact  
(Wikipedia page on 'Calcite')

Both rhombohedron and scalenohedron are closed form - faces enclose space completely

General form  $\{h\bar{k}\bar{l}\}$  is a scalenohedron, 12 scalene triangles

Common forms for rhombohedron:  $\{h0\bar{h}\}$  and  $\{0h\bar{h}\}$   
- 6 rhomb faces, like squished or elongated cube

Common rhombohedron index is  $\{10\bar{1}\}$ .

(Earth Materials, Chapter 4 + 5)

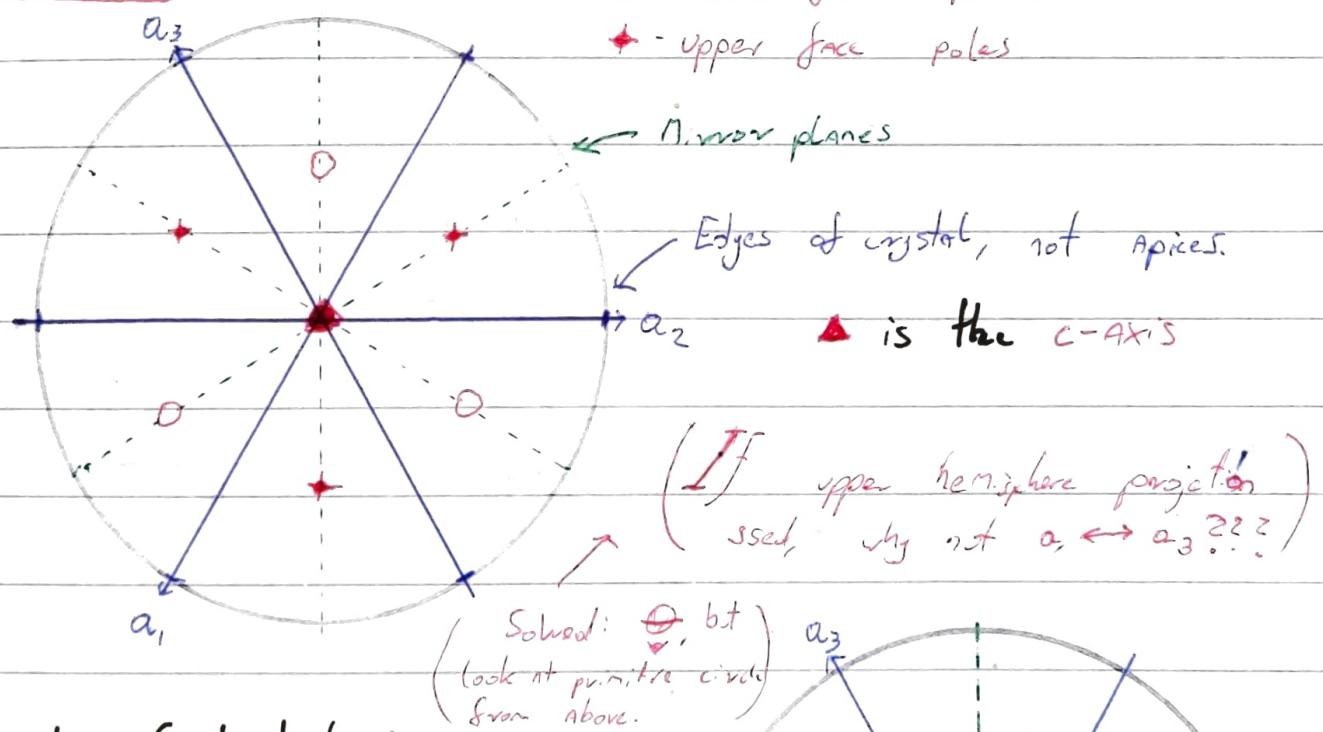
## Stereonet projections:

In  $\bar{3}2/m$ , c-axis is the not inversion axis.

Horizontal axes are 2-fold rotation

In the rhombohedron form, (and maybe scalenohedron too), parallel faces are by inversion, NOT reflection

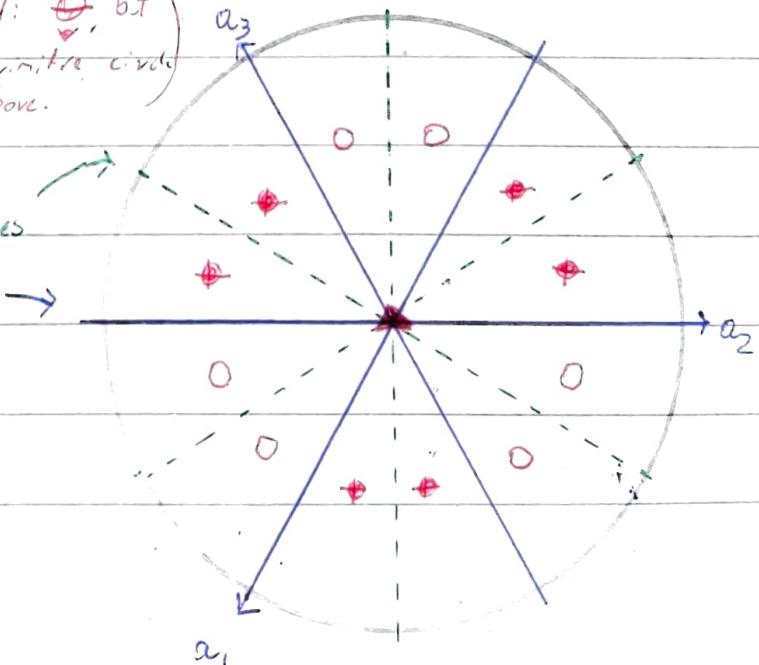
### Rhombohedron:



### Hexagonal Scalenohedron:

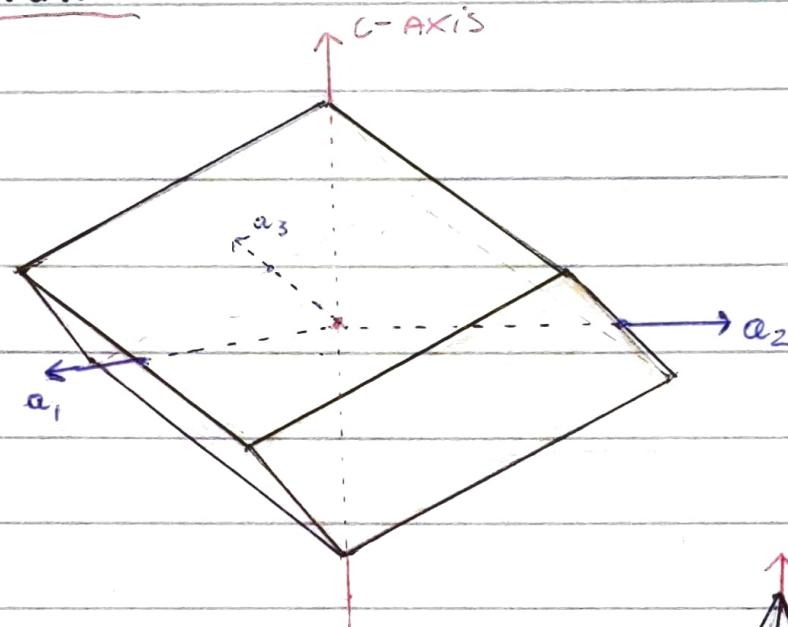
pretty sure also the EDGES,  $\rightarrow$   
not apices.

O - lower face projections  
F - upper face projections

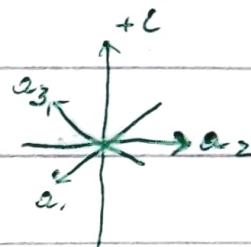


## Two common forms of Calotte drawings

### Rhombohedron:

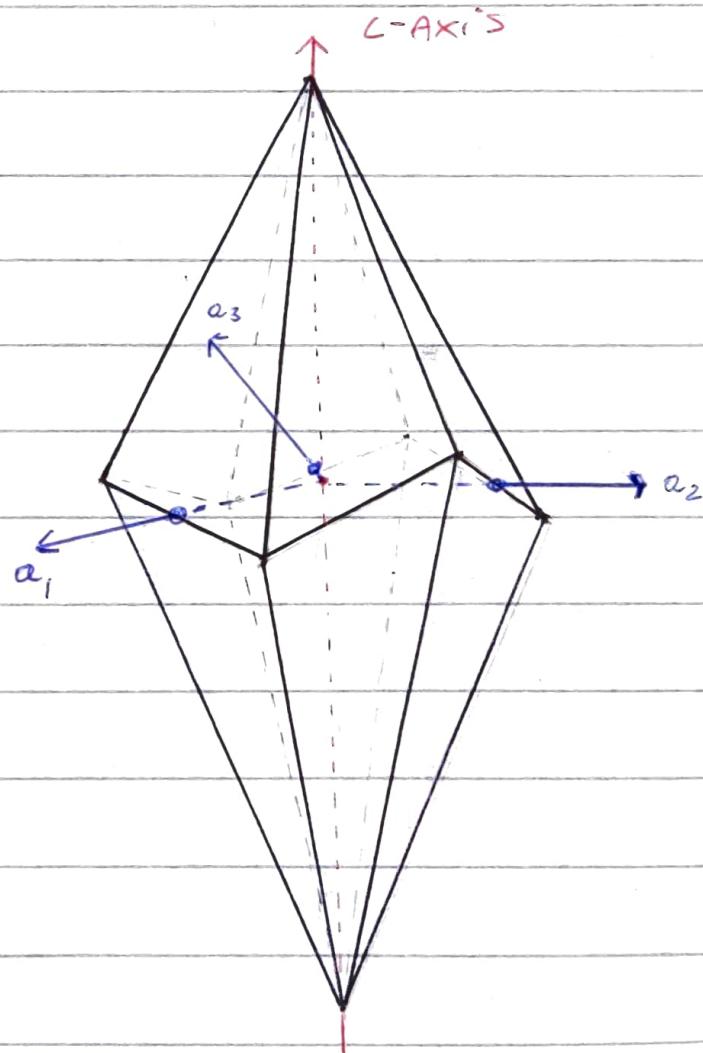


(clinoigraphic projections)



### Hexagonal scalenohedron:

- Both have C-axis exiting through an apex
- Both have Horizontal axes exiting through edges





Twin element  $\rightarrow$  New shared symmetry element

- Mirror plane or twin plane
  - Rotation axis or twin axis
  - Inversion about a point or twin centre
- } Twin laws

For Growth twins + Deformation is not a growth twin?

Interested in Secondary twinning

Classification of twins:

Contact twin  $\rightarrow$  separated by twin plane

Penetrating twins  $\rightarrow$  Appear Intergrown  $\rightarrow$  Axis of rotation common

Multiple  $\rightarrow$  3 or more crystals twinned, accords to 2 laws

Polysynthetic twin - All composition surfaces are parallel

Ref of chapter 5 if needed  $\Rightarrow$  Bravais notation



## 1st Lecture with Enric:

Dynamic recrystallization: Create recrystalline at high stresses  
Bigger stress  $\rightarrow$  smaller grains

c-axis  $\rightarrow$  trigonal notation axis

Trigonal system  $\rightarrow$  holosymmetric class

No centre of symmetry  $\rightarrow$  piezoelectric properties (Qtz)

Calcite has centre of symmetry  $\rightarrow$  no piezoelectric properties

## Nomenclature for calcite (carbonate mineral)

→ faces called  $(v_1, v_2, v_3)$

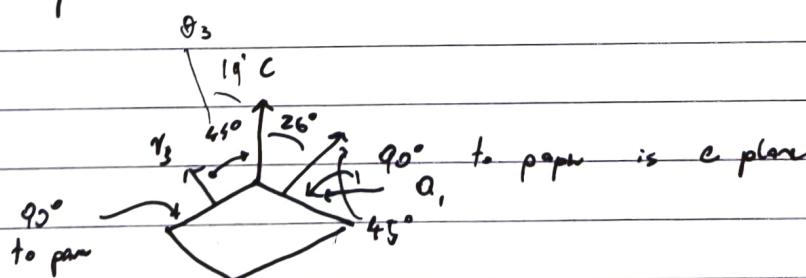
→ steep faces (spur, fayal) ( $f_1, f_2, f_3$ )

→ basal (c)

a plane  $\rightarrow$  halfway between  $v_1$  and  $v_2$   
parallel to ~~opposite~~ edge  
twining plane

Each parallel to Edge between r planes

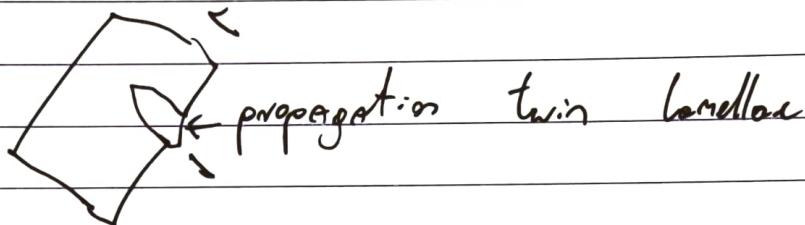
3 c planes



Position sense of twining  $\rightarrow$  moves upper towards C-AXIS.  
THE Only possible sense of twining

(2)

Shear can produce twinning - Martensitic transformation



In calcite only 3 slip systems ( $e_1, e_2, e_3$ )

Need 5 to satisfy Taylor - "phenomeno" <sup>processes?</sup>  $\rightarrow$  need more planes

Smaller crystal harder to twin as backstress (from surface energy) fights against it.

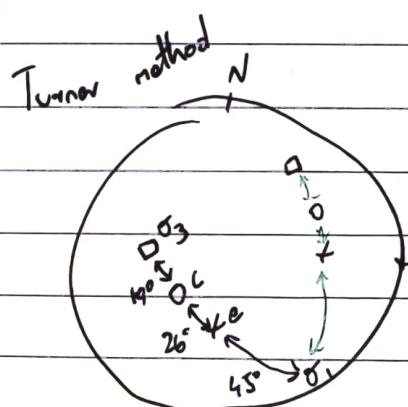
Small enough & grains don't twin

- Shearing in opposite sense is not possible.  
Will de-twin already twinned calcite.

Optimum direction of compression at  $45^\circ$

If see a twin  $\rightarrow$  assume optimally oriented

Plot this on stereonet will give directions of  $e_1$  &  $e_2$



Multiple sets of grains can tell  $e_1$

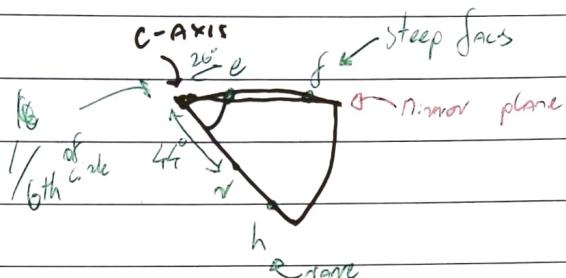
• Think about good refraction from pole figure

(3)

On Ermis' paper:  
(Rutter, 1994)

+ Asg + Bsdm

Or relate geographic axes to crystal axes ↗ "inverse pole figure"



To link crystallographic axes to outside world

- 9 direction cosines, 6 of which independent
- 3 Euler angles ( $0 - 180^\circ$ )

Or s should cluster around a point

From cluster  $\rightarrow$  harmonic spherical function fit  
"Orientation distribution function"

Inverse-pole figure is a section of ODF.

↙ back-calculate data to check GoF

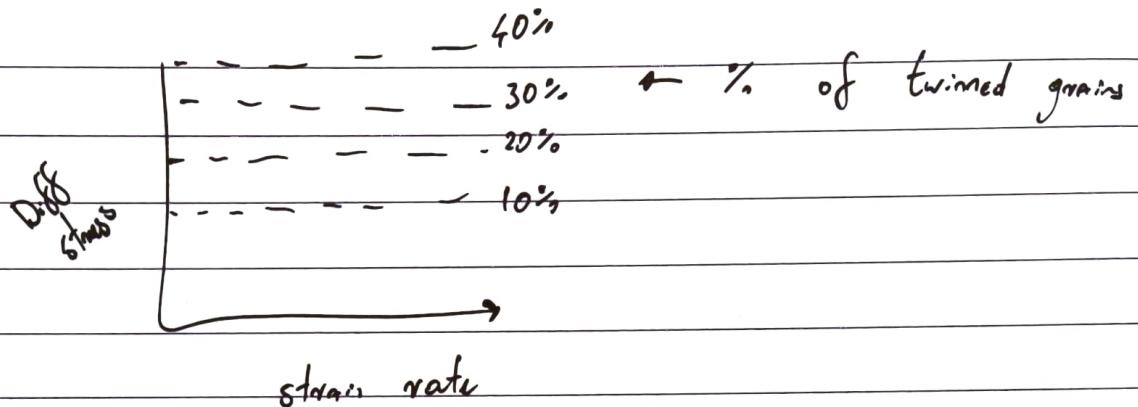
At higher temperature natural weaker  $\rightarrow$

E-maxim fabric  $\rightarrow$  low temp, twining dominated

E+Ah fabric  $\rightarrow$  not to do with twining ↗ Sudden flip  
at certain conditions

Twining Saves in compression

(4)



## • FIND PLANAR GRAIN SIZE DISTRIBUTION

2 Problems with planar grain size:

- Determination of orientation of  $\sigma_1, \sigma_2, \sigma_3$
- Determination of magnitudes.

• Twin boundary migration can have some effect.

→ In EBSD → Test for twin:

72 degrees between crystallography and a twin?  
 ↳ ~~specz~~

Twins + slips on ref may cause rotations in host grain  
 <10° variation for heterogeneous strain

Aim: Extract a good parameter describing twin activity

- eg.
- Twin density
- Twin number
- Thickness (etc.)

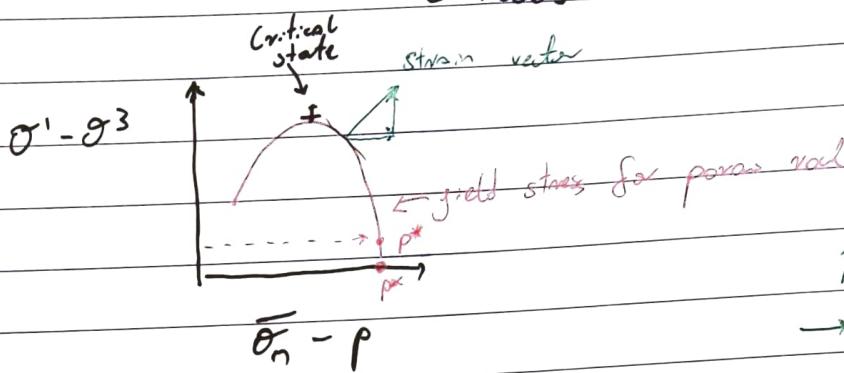
Schmid factor:

- Does not work well for twinning because of grain size effects on twinning stress

Aim

Extract good Palaeoplasticity

(5)

Yield stresses  $\rightarrow$  Crystal-plast.  $\rightarrow$  $\rightarrow$  2 models

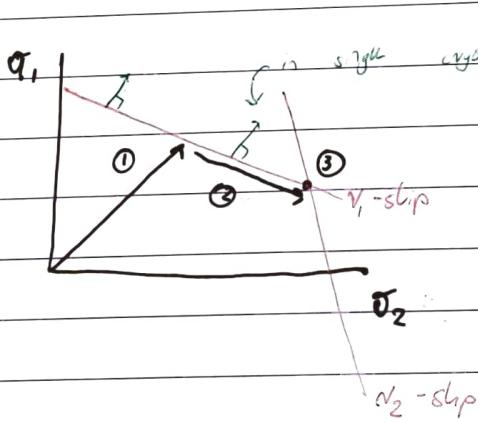
Strain vector normal to failure

extens.

 $\uparrow$  Distortion strain $\rightarrow$  Volumetric strainBut often Strain vector not  $\perp$  to failure criterion

"Non-associated flow"

For single crystal:

① Probe approaches  $n_1$ , and failure initiates

② Strain rotates mineral relative to others

③ Equal stress on both mineral slip systems

④ In 3D, travel long edge to a vertex

⑤ And so on into 5-dimensional space

(Taylor, Bishop, Hill analysis)

Carrara Marble  $\rightarrow$  Relatively free of Twins to begin with  
1978 paper  $\leftarrow$  5 dimensional stress

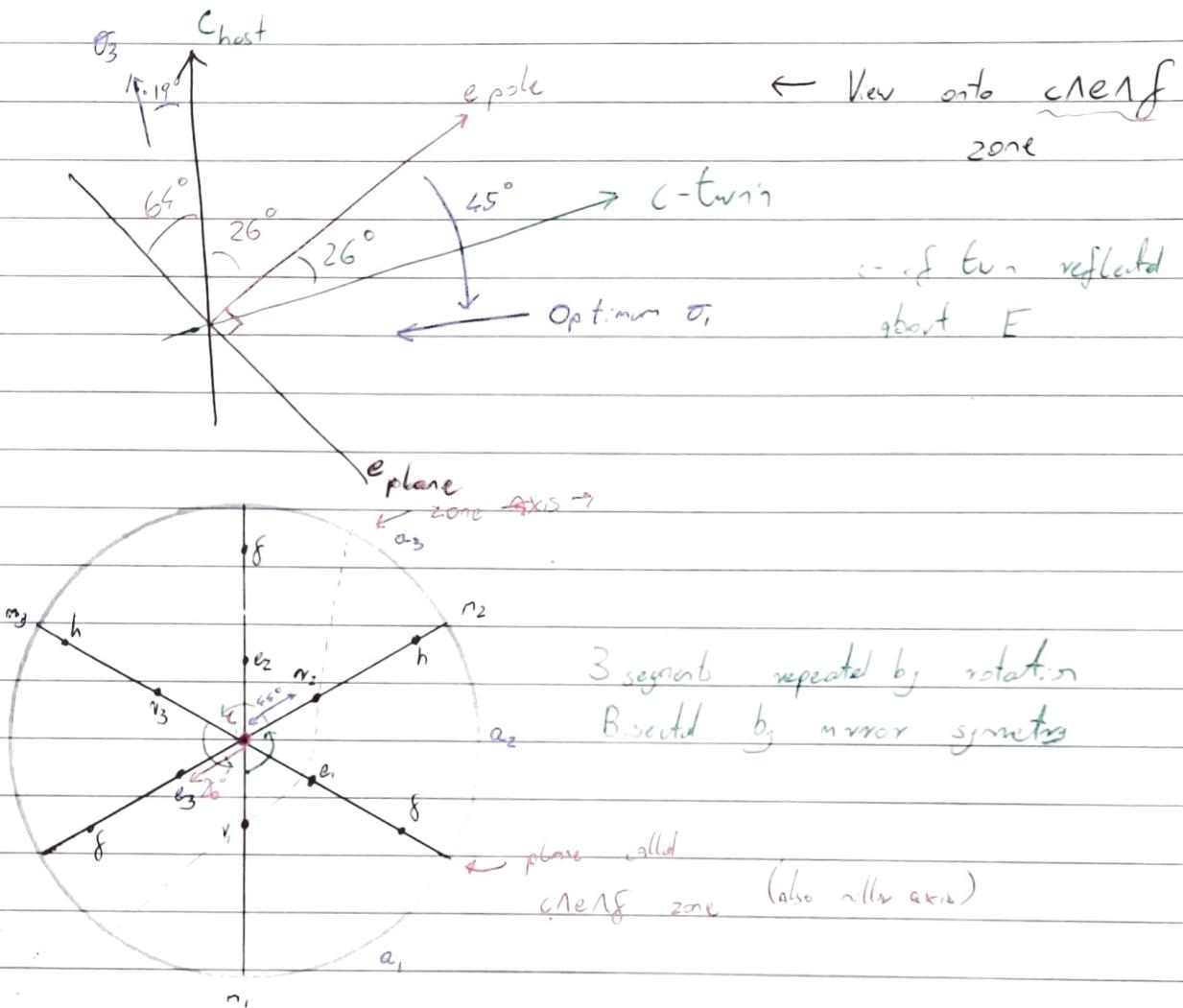
Burster William textbook too

PLAY WITH DATA!

①

Second Meet. w. H. Erne.

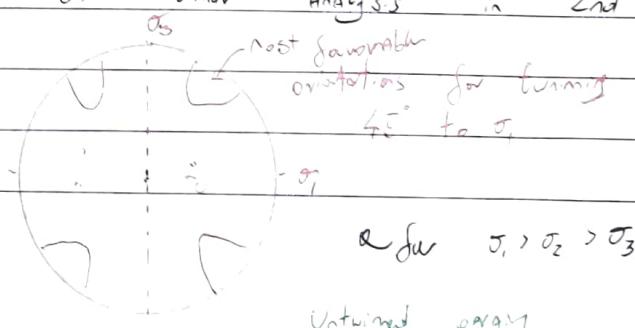
Try MTEX for all plotting



Str. deformation important for fabric/defor. at high temp.  
At low temp  $\rightarrow$  stay in twinning mode

Use WEDS (equal angle net)

More on twin analysis in 2nd lecture



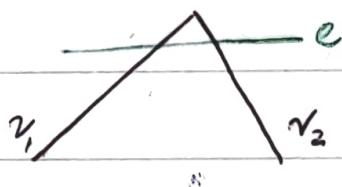
## Few more notes about Calcite

(based on both meetings with Ernie)

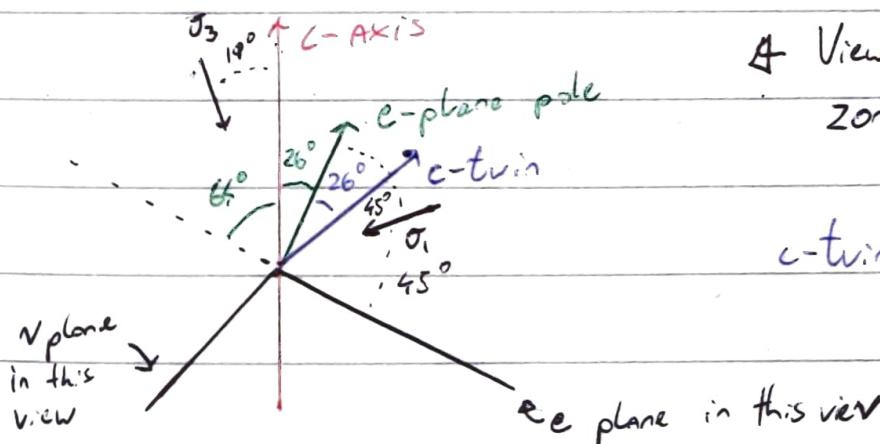
Faces and other planes have particular nomenclature:

- faces are called " $r_1, r_2, r_3$ "
- steep faces are " $f_1, f_2, f_3$ "
- basal faces are " $c$ "

" $c$ " plane is a twinning plane, which lies halfway between two " $r$ " planes:



$c$ -plane is parallel to non-a edges in rhombohedron.



A View onto the  $c\wedge c\wedge c$  zone

c-twin effectively reflected around  $c$ -plane.

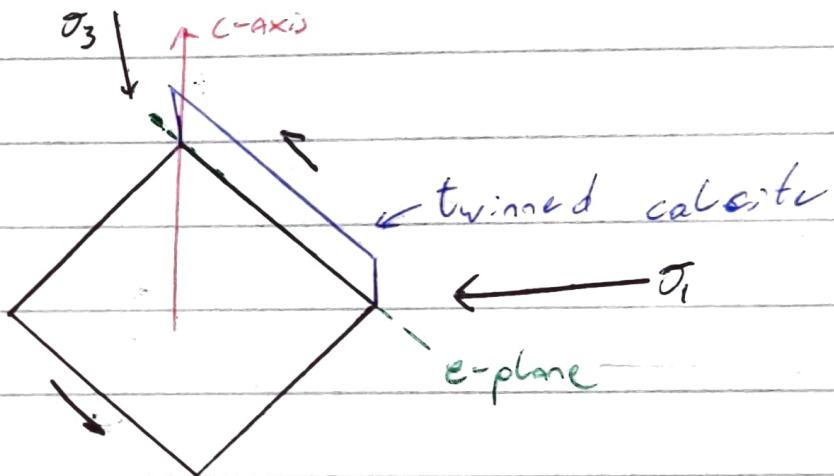
both  $\perp$  to page

$\sigma_1$  is at  $45^\circ$  to  $c$  plane.

05 MAR 2021

Notes on calcite continued:

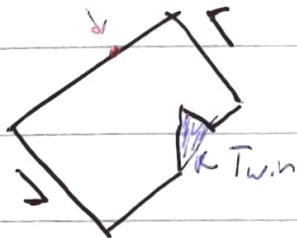
- Calcite twinning allows for only 3 slip-systems. That is, there are only 3 c-planes.
- Need 5 to satisfy 'von Mises' criterion (if no volume change).
- Hence some slip must occur on  $\text{W}$  or  $\text{F}$  or other planes.



- Only positive sense of shear is possible in calcite (Towards the c-axis).
- In the other sense of shear, calcite will not twin, and even de-twin if already twinned!

Hence, if see a twin, assume it is optimally oriented!

## Twinning in Calcite



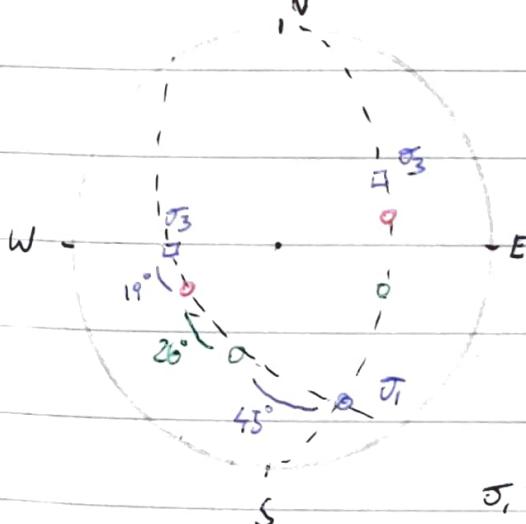
- Tens propagate through the crystal as "wedges", but are opposed by the backstress caused by the surface energy at the other side of the crystal.

Hence, bigger crystals twin much easier than smaller crystals.

At high strain rates, calcite undergoes "dynamic recrystallization", which reduces the grain size and potentially removes twins.

## Turner Method

Turner method tells us the  $\sigma_1$  and  $\sigma_2$  directions.



- ① Plot  $c$ -axis for <sup>few</sup> host grains
  - ② Plot e-poles for those grains
  - ③ Plot  $\sigma_1$   $45^\circ$  from e and  $\sigma_3$   $19^\circ$  away from c.

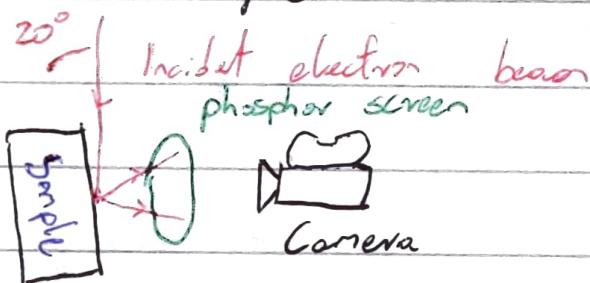
σ, - should cluster in one area.

$J_3$  may be spread if stars axi-symmetric

08 MAR 2021

## Notes on EBSD

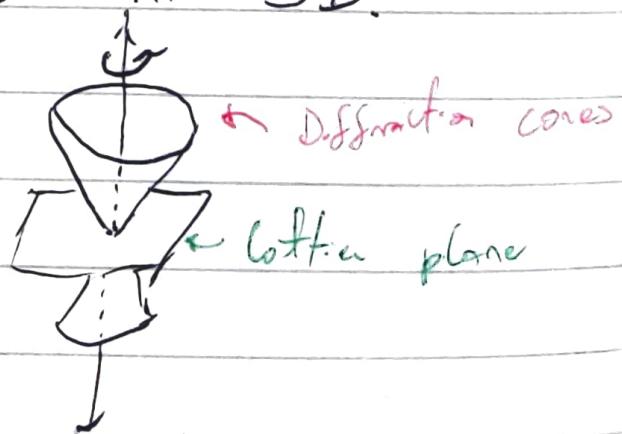
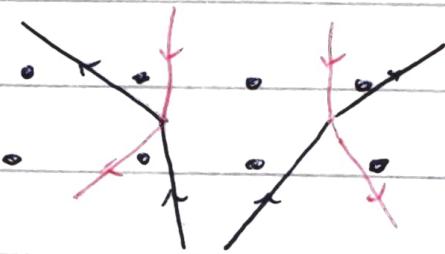
- Electron back-scatter diffraction gives us the orientation of a crystal within a sample



1. Electron beam hits the sample at a shallow angle

- 2. Interaction within upper  $\mu\text{m's}$  of the sample create an omnidirectional source of electrons

- 3. Electrons are diffracted by the near-surface lattice planes. There are 4 possible paths satisfying the Bragg equation in 2D, which translate to two cones in 3D.



\* Neat to be symmetrical

## Notes on EBSD continued

4. At the electron energies typically used in EBSD, the diffraction angles satisfying the Bragg equation,

$$n\lambda = 2d \sin \theta,$$

where  $n$  is the order of diffraction,

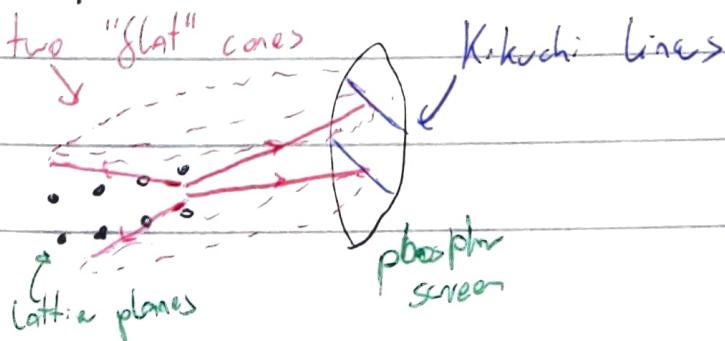
$\lambda$  is the electron wavelength

$d$  is the lattice spacing

and  $\theta$  is the Bragg angle

The two are typically pretty shallow. In fact, they are almost  $90^\circ$  and the two cones are open at about  $180^\circ$

5. The two almost-parallel planes hit a phosphor screen and produce a Kikuchi pattern



Notes on EBSD continued

6. The image produced on the phosphor screen is captured by a CCD camera.
7. The bisectors of individual K.Kuchi bands are the projections of lattice planes.
8. Computer uses Hough transforms to process the data and index (compare) the pattern to a known data base.
9. Once orientation is known, process is repeated at the next site, eventually producing a map of crystal orientations.

The process usually takes place within an adapted Scanning Electron Microscope (SEM)

Ern's 1989 paper notes

Found link between : Diff stress and twinning induce  
 Density + Volume fraction  
 Grain size control

Verify Ern's data in my findings

Turner 1953 for Turner method  
 (Turner Dynamic Analysis) - TDA

Good review by Turner & Weiss 1963

Testing apparatus might be ~~softened~~ by described by Ritter 1972 or 1985

Low ( $<0.2\%$ ) porosity

Carriers relatively untwinned

Induced twins tend to be benticular

Results based on 200-300 prms

Twinning largely independent of strain rate and Temp but  
 Sensitive to stress

Not thermally  
 activated. Ideal  
 indicator

$\sim 20\%$  of grains expected to not show  
 twinning - unfavorable orientation  
 for Polars

for Polars  
 indicator

Twin density  $\propto$  independent of Grain size

$\propto$  dependent on stress

Schmid factor,  $m$

$$m = \cos \phi \cos \psi$$

$\phi \rightarrow$  Angle between

Normal between ~~maximum compression~~

maximum compression direction and normal to twin plane

$\psi \rightarrow$  Angle between compression direction and turning displacement vector

Residual shear stress remains when both at  $45^\circ$

$$0 < m < 0.5$$

Volume of twin analogs is independent of deformation history

For it is good analogue to natural rocks

Volume, hardness and Density all useful

For simple history, more accurate

Emri's B paper

Crystallographic Orientation development during deformation

Slow deformation is good because stress is always homogeneous

In compression, stress not necessarily proportional to cross-sectional area

At  $300-500^{\circ}\text{C}$  flow is strain-hardening

At low temp,  $<600^{\circ}\text{C}$  flow insensitive to grain size

At mid temp, mid stress,  $600-700^{\circ}\text{C}$  flow becomes grain size sensitive

Twinning easier at larger grain size

→ Natural rocks almost always show e-fabric?  
(low temp)

Inputs strongly inhibit calcite recrystallization

Twinned calcite rock ~~can~~ anneal back to equilibrium at high temperatures & twinning gone but CPO present

CPO fabrics can survive subsequent deformation

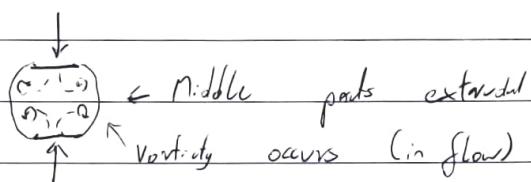
Intracategorical Plasticity (a) is what interests me  
(e.g. fabric)

## Distribution of non-plane strain - Schiøtz

- Short cylinders (radius > length) length half the radius
  - ↳ Non-plane strain  $\rightarrow$  heterogeneities present

Crystal plasticity  $\rightarrow$  deformation mechanism

SPO  $\rightarrow$  Shape preferred orientation



Specimen diameter  $\rightarrow$  9.5 mm

Good diagram of pizza slice with Miller-Bravais indices on it

In general, shows CPO from EBSD data transect across a sample deformed in multiple regimes

Still no e+alpha in natural rocks

Very crude notes on Evans '89 paper

→ There is a link between Differential Pressure and:

- Twining incidence (Also found that there is a grain size effect - small grains don't twin)
- Twining density
- Twin Volume fraction

(I should confirm the same relationships in my data)

→ Twining largely independent of strain rate and temperature, meaning twining is not thermally activated ← great paleogeometer

→ Only favourably oriented grains twin. About 20% of grains expected to not show twining as they are unfavorably oriented.

→ Induced twins tend to be more lenticular

→ Twin density is independent of grain size, but still dependent on stress

→ Twin volume fraction analysis is independent of deformational history

15 MAR 2021

Crude notes on Eric's '93 paper

- Lists advantages of deformation in extension, but those do not apply to my study.
- Twining is easier at larger grain sizes.
- At 300-500°C, flow is strain-hardening. Low temperature is important, as that's where the e-fabri. develops
- Natural rocks show e-fabri. only
- Twinned calcite rocks anneal back to equigranular structure at high temperature. Twining does appear, but CPO remains
- Those CPO fabrics can survive subsequent deformation
- Intracrystalline plasticity is the term from this paper that interests me. It leads to e-fabri.

## Rutter 14

- Because most minerals are seismically anisotropic, a rock with a CPO should too show an seismic anisotropy
- Materials 5 slip systems
- Crystal lattice rotates (along with slip system) as slip occurs until at "easy slip" + steady condition, easiest slip
- Conjugate shear systems can cause asynchrony
- Can somehow use this asynchrony in shear analysis
- 3D analysis of various parts in rock should give good image of stresses in rock + whatever the deformation regime each part expands

Once again, impurities prevent recrystallisation  $\rightarrow$  Allows deformation at higher temperatures than otherwise possible.

$\hookrightarrow$  High possibility of more slip systems + grain boundary sliding

$\rightarrow$  Interesting seismic properties of calcite

$\rightarrow$  Shaded CPO has an effect on seismic anisotropy

Extension  $\rightarrow$  High confining pressure and withdrawal pressure

$\rightarrow$  Conditions such that Saffman deforms in intracrystalline plasticity regime

Truncating in some larger grains

Seismic properties calculated, not observed?

$h$  calcite,  $V_p$  lowest parallel to  $c$ -axis

- Use intensity of poles to f-planes to measure the strength of CPO  $\rightarrow$  but for extension?

$\rightarrow$  Talks about how to estimate strain in compression

$\rightarrow$  Intens. does not increase systematically with shortening

Kern 1977 1979 + Shorai CPO in Limestones

- Neither patterns or their intensities correspond to nature  
e.g. in nature, crystallization operates

Links to Rutter 1976, 1990, 2007

NOT ABOUT TWINNING IN THIS PAPER

Don't think this paper relates much to what I am doing except on the f-plane IPF  $\rightarrow$  strength CPO C.V.

In nature, Calcite provides non slip surfaces the other rock types mind

Ruttar → Karakoram 2007

- Karakoram fault ~~has~~ has a mylonite which shows ductile deformation (intracrystalline plasticity)

- Only 40-150 km

- Of interest here → Pangong range cutting through Karakoram terrane → Deep at crustal rocks

3m/year on strike/ slip

Quaternary slip rate on Pangong estimated at 4mm/year

Low seismicity and low recurrence rates

- 15M<sub>a</sub> leucogranites sheared & faulted younger than that

Paper tries to establish P-T history

Calcite (Murb) exposed at Murb

USED TURNER Analysis to produce back ball diagrams

- Can I do this for my analysis?

Ask Emi what Emi did

Turner only in coarser grains

Turner can tell pressure, provided the event did not cause recrystallisation

Ask Emi

→ CPO develops c-axis  $\perp$  to foliation

→ Common fabric for shear and mineral stretching

→ Sample 6km from fault trace has all-grains twinned

→ No dynamic recrystallisation found

→ Catastrophic fragmentation observed by CPO

Inclusions: Transpressive → No surprise →  $11\backslash\backslash\backslash\backslash\backslash$

Grain-size sensitive shear by Walker 1990 → Might be important

Anchizone — Epizone???

Diagonistics → Metamorphic core

Steve and Eric worked on Naxos, Greece marble 1989

Good:

→ Twining good despite  $\approx 30 \text{ MPa}$  confining, as strain rate and temp independent

BUT GRAIN SIZE IMPORTANT

→ Fig 8

Links slip zone width to slip rate

Vibr zone = high slip rate

## Brief Notes on Ernie's '07 Karakorum paper

- Example of a practical application of Calcite palaeoporosity
- Used Turner analysis on marbles in a wide shear zone to produce several "beach-ball" diagrams mapping the shear zone
  - perhaps I can try to do this for my analysis?
- Method only works if rocks did not dynamically recrystallise - Ernie thinks this is an example of such rocks
- Found that Calcite is a good palaeostress indicator despite 30 MPa error bars, as it is strain rate and temperature independent.
- But as always, it is grain-size sensitive and that must be taken into account.
  - Probably need to produce grain-size distributions for my analysis.

19 MAR 2021

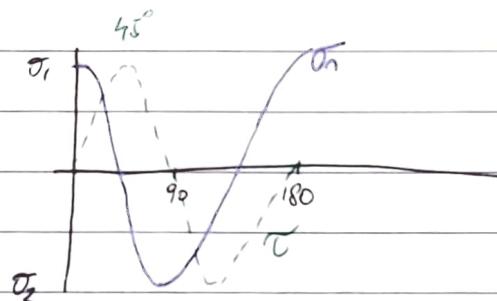
Brief notes on Ernie's '14 paper

- Looks at seismic anisotropy in Calcite crystals, and calcite rocks with a CPO
- c-axis is the slowest seismic direction in Calcite
- Perhaps a useful link between poles to S-planes and IPF to provide a measure of CPO strength
  - Not sure if applies to my "type" of deformation though
- In general, I don't think this paper relates much to what I am doing.

# Geomechanics for petroleum engineers

Heterogeneous stress  $\rightarrow$  Stress trajectory orientation vary from place to place  
If they converge, component increases in value

Shear stress numerically at maximum at  $45^\circ$  to principal stress



$$\sigma_n = \frac{1}{2}(\sigma_1 + \sigma_2) + \frac{1}{2}(\sigma_1 - \sigma_2) \cos(2\theta)$$

$$\tau = \frac{1}{2}(\sigma_1 - \sigma_2) \sin(2\theta)$$

$\tau$  is the normal of a plane  $\perp$  normal to  $\sigma_1, \sigma_2, \sigma_3$

Homogeneous Strain  $\rightarrow$  Straight lines remain straight, parallel lines remain parallel, but angles change

Young's modulus :  $E = \frac{\text{stress}}{\text{strain}} = \frac{\sigma}{\epsilon}$

Poisson ratio :  $\nu = \frac{\epsilon_y}{\epsilon_x}$

Individual strains :  $\epsilon_1 = \frac{\sigma_1}{E} - \nu \frac{\sigma_2}{E} - \nu \frac{\sigma_3}{E}$

$$\epsilon_2 = \frac{\sigma_2}{E} - \nu \frac{\sigma_1}{E} - \nu \frac{\sigma_3}{E}$$

$$\epsilon_3 = \frac{\sigma_3}{E} - \nu \frac{\sigma_1}{E} - \nu \frac{\sigma_2}{E}$$

(2)

Wave velocities:

$$V_p = \left( k + \left( \frac{G_p}{3} \right) \rho \right)^{1/2}$$

$$V = [\mu/\rho]^{1/2}$$

K is the bulk modulus:

Roughly

$$K = \frac{\partial \sigma}{\partial \epsilon_v}$$

volume

$$K = \frac{E}{3(1-2\nu)}$$

Poisson's

Young's modulus

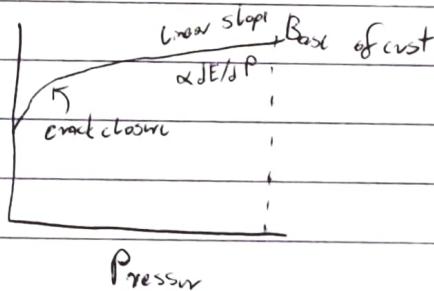
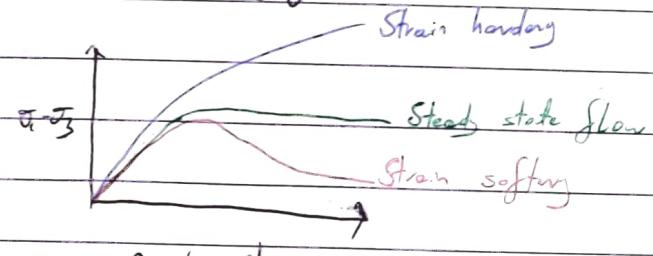
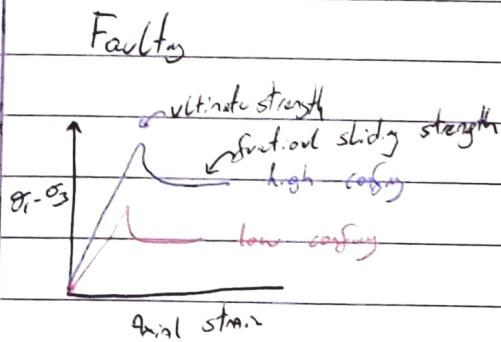
 $\mu$  is the shear modulus (G)

$$\mu = G = \frac{\partial \tau_{ij}}{\partial \gamma_{ij}}$$

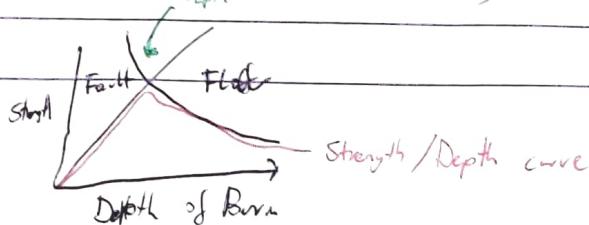
$$= \frac{E}{2(1+\nu)}$$

Poisson's

Young's modulus

Rocks begin to flow above  $300^\circ C$  (salt at  $80^\circ C$ !)

h crust: Depth limit of seismicity



(3)

Creation of cracks requires volume increase. This becomes harder with depth because of increasing confining pressure. Shear stress must be high to overcome that. Pressure insensitive.

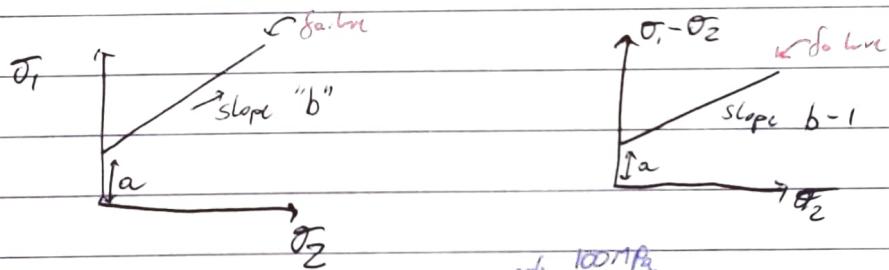
At greater depths temperatures increase and plastic flow becomes easier. Plastic flow preserves volume and so is pressure insensitive.

One principal stress can be expected to be perpendicular to ground surface as  $\sigma = 0$

Faults at  $30^\circ$  to  $\sigma_1$

$\sigma_2$  lies in fault plane

Mohr - Coulomb failure for low porosity rocks in upper crust



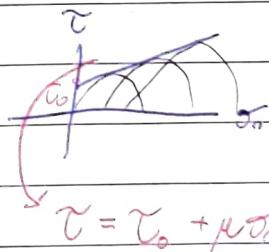
Usually a straight line  $\rightarrow a + b\sigma_2 = \sigma_1$

$$a + b\sigma_2 = \sigma_1$$

(cohesion 100 MPa)  $\rightarrow$  (cohesion 4)

Unloading  $\rightarrow 0$  for sliding on existing fault

Those plotted on Mohr circles make a "Mohr envelope"  $\rightarrow$  Strength for intact rocks



$\sigma_c$  is the cohesive strength (intercept)  
 $\mu$  is the coefficient of friction (gradient)

$$\tau = \sigma_c + \mu \sigma_n$$

(4)

For transformations:

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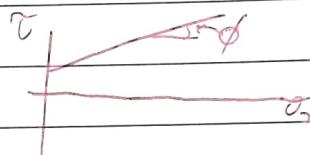
$$a = 2C_0 \sqrt{b}$$

$$b = \frac{(1 + \sin \phi)}{(1 - \sin \phi)}$$

$$\mu = \tan \phi$$

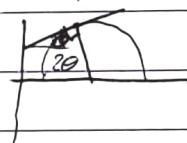
$\phi$  slope of the envelope from horizontal.

Usually  $\sim 30^\circ$



Angle of fault to  $\sigma_1$  is  $\theta = 45 - \frac{\phi}{2}$ :

Geometry works out



Pore fluid pressure brings circle to the left

$$\text{a: } \tau = C_0 \mu (\sigma_1 - p) \quad \text{pore fluid pressure}$$

On overpressure:

$$\lambda = \frac{p_{\text{fluid}}}{p_{\text{rock}}} = 0.4 \text{ if pores connected to surface}$$

$\mu$  tends to be 0.6 to 0.75  
e.g. Biot's rule

Hydraulic fractures propagate in plane normal to  $\sigma_3$

Mohr-Coulomb requires work to be done to create 1-2% dilation

↳ Hence rocks get stronger with depth.

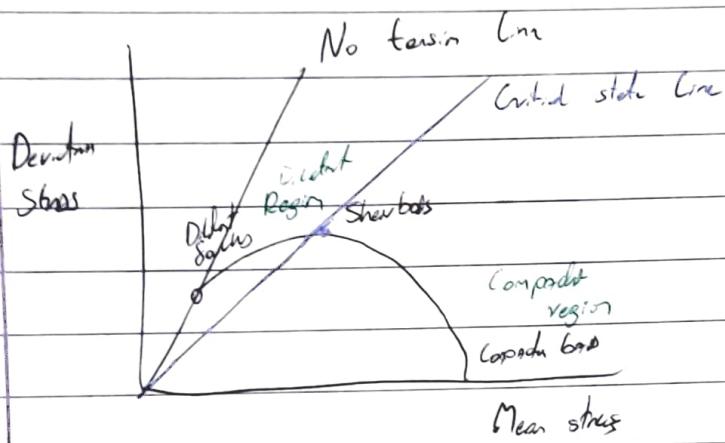
→ Works for porosities less than 10% → Otherwise more complex

(5)

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## Anhennius State Law:

$$\text{Rate} \propto e^{(-H/RT)} \quad \text{Not sure what } H \text{ is}$$



Poisson ratio of  $\approx 0.5$  assume viscous fluid

Finišank Near Kirsch equation

# Geomechanics for Pet Eng PDF

a) Body forces  $\rightarrow$  Force per unit volume

b) Surface forces  $\rightarrow$  per unit area  $\rightarrow$  force or Stress or Pressure

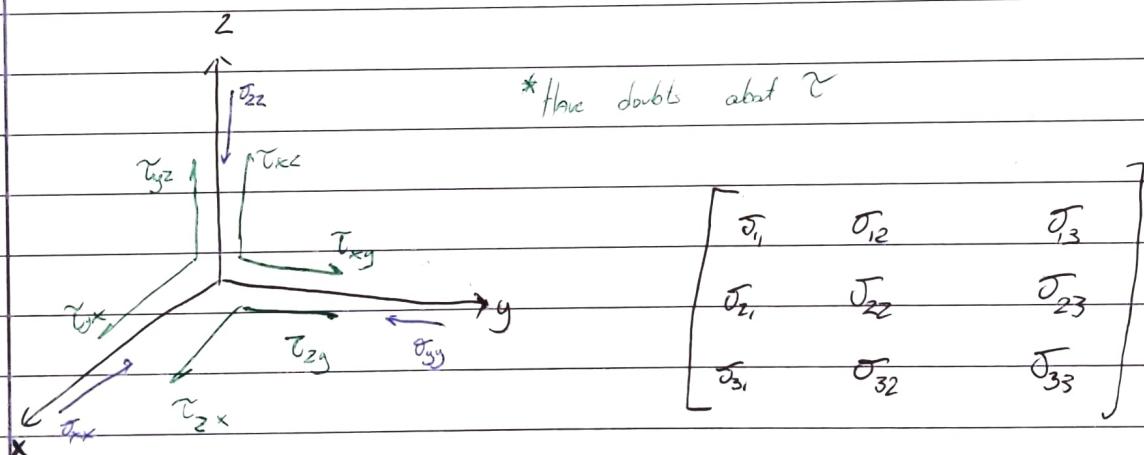
Conventions:

Normal stress  $\rightarrow$  Compression  $\rightarrow$  Positive

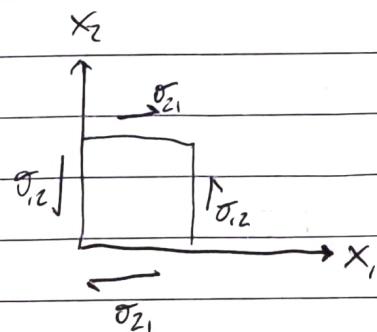
Shear stress  $\rightarrow$  Right-handed  $\rightarrow$  Positive

In 3D:

- First subscript is the Normal to the plane across which component acts
- Second subscript is the direction in which the component acts



Stress is a tensor quantity



$$\tau_{ij} = -\tau_{ji} \leftarrow \text{Symmetric about leading diagonal}$$

(2)

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(3)

Always possible to find orientation so that shear stresses are  $= 0$

Orthogonal grid = "Stress trajectory"

$$\sigma_n = \frac{1}{2}(\sigma_1 + \sigma_2) \pm \frac{1}{2}(\sigma_1 - \sigma_2) \cos(2\theta)$$

$$\tau = \pm \frac{1}{2}(\sigma_1 - \sigma_2) \sin(2\theta)$$

Basically parabolic equation for a circle, hence Note

$\theta$  is angle of place to

$\tau$  at max when  $\sin 2\theta = 1$ , or  $\theta = 45^\circ$

Parabolic equation to a circle

$$x = C + r \cos \alpha$$

$$y = \pm r \sin \alpha$$

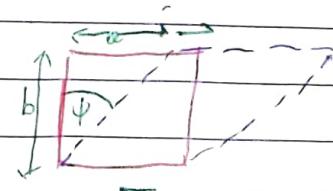
Strain:

$$\text{Strain of line: } e = \frac{\Delta L}{L_0}$$

$$\text{Quadratic elongation: } \lambda = (L/L_0)^2 = (1+e)^2$$

$$\text{Natural or logarithmic strain } \epsilon = \ln(1+e)$$

$$\text{Shear strain, } \gamma = a/b = \tan \psi$$



Elastically up to 1% strain

### ③ Elastic moduli

$$\text{Young's Modulus } E = \sigma_x / \epsilon_x$$

$$\text{Shear modulus } G = \sigma_{xy} / \epsilon_{xy}$$

$$\text{Bulk modulus } K = \sigma_{xx} / \epsilon_{xx}$$

$$\text{Poisson's ratio } \nu = E_x / \epsilon_y \quad \text{for load applied in y direction}$$

$\nu = 0.5$  = isovolumetric

for rocks  $\nu \approx 0.3$   $\leftarrow$  Assumes volume reduction

P waves twice as fast as S-waves

$\approx 6.5 \text{ km/s}$

$$\rho \approx 2700 \text{ kg/m}^3$$

Typical pressure gradient:  $35 \text{ MPa/km}$

At Moho  $\rightarrow 16 \text{ GPa}$

Intracrystalline plasticity: Slippage of atomic planes  
one over another

04 APR 2021

Simple shear, constant volume. Pressure insensitive  
 $\hookrightarrow$  surely not?

Most common deformation mechanism in lower crust

$\hookrightarrow$  at shallower depth salt  
does this

Dynamic recrystallization starts around grain edges



$\hookrightarrow$  Replaces strained structure with unstrained structure

(4)

Higher stress = Smaller grain when recrystallized

What is Von Mises / Taylor Condition?

Stress homogeneous

2 poles are  $(01\bar{1}2)$

Fabric forms as twin slip on  $\epsilon$  and  $\tau$   $\checkmark$  slip on

Host grain c-axes rotated towards  $\tau$

c-axes already at near  $110^\circ$ , migrate to  
 $20-30^\circ$  small angle  
 Also no twin - link band  
 $\hookrightarrow$  slip on  $\tau$  instead

Shear strain for twinning

$$\gamma_t = k V_f$$

(Volume fraction of twin)

0.69% for calcite + Shear strain for complete twinning

Intensity Sh. factor for twinning grps

Twining so easy, we don't need 5 robots as ~~XX~~ theorized

Elas. Angles  $\rightarrow$

$m \rightarrow x$

$E_3 ?$

$a \rightarrow y$

$E_2 ?$

$c \rightarrow z$

$E_1 ?$

Clawwise is default  $\rightarrow$  lateral anticlinal.

Restorative clawwise

Buys:  $2 \times 2$

$\psi$  About  $z$  Axis,  $-180^\circ$  to  $180^\circ$

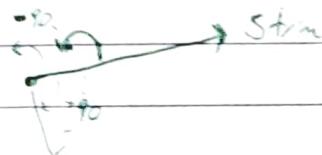
$\theta$  About  $x$  Axis,  $0$  to  $180^\circ$

$\psi_2$  About  $z$  Axis

They are rotations which bring lab in concordance with crystal

Try again:  $E_1$  = Azimuth strike (but may be  $90^\circ$  off from bygram)

$E_2$  = Plunge (now or dip) but if now then  $\cancel{45^\circ}, 90^\circ$ , flip azimuth by  $180^\circ$



$$E \\ 180 - 90 - 180, Azimuth = 270 + (180 - E)$$

$$Wth E, \text{Geographic } 0-90, Azimuth = 90 - E$$

Actually this is a "strike"

$$180 \\ 270 \\ 90 \\ 0 \\ 90$$

$$180 \\ 90 \\ 0 \\ 90$$

Bore

E

$$If 90 - 180, Azimuth = 0 - (-90)$$

Never get in this zone initially.

Will be later when  $E_2 > 90$

Ex, if  $E_2 > 90$ , Azimuth gets extra  $+90^\circ$

Then if  $> 360$ ,  $\text{Do } -360$

the

$$\text{Plane: If over } 90, d_p = 180 - E_2 \\ \text{Else } d_p = E_2$$

if  $0-90$

Azimuth:

Azimuth begins here

↓ now impossible for the Azimuth



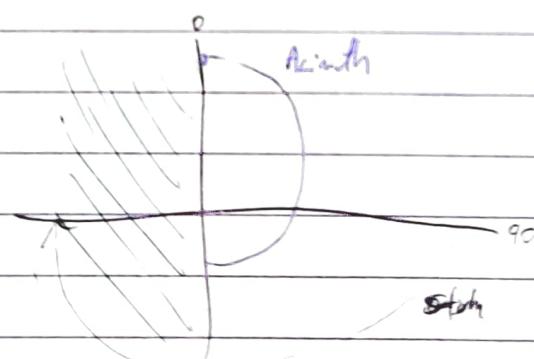
$360 - E$

$0, 90 \rightarrow 90 - E, 270$

Perhaps only other way?  
 $90, 180$

$270 - E$

Treat rotation as clockwise



$0 - 90 = E$

## Schmid law / criterion / factor

- Describes the slip plane and slip direction of a stressed material which can resist the most shear stress

Critically resolved shear stress,  $\tau$

$$\tau = n \sigma$$

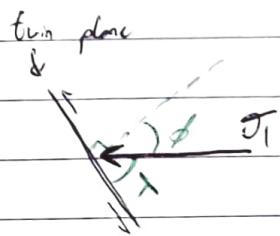
↗ ↘  
 Stress applied  
 Schmid factor

$$n = \cos(\phi) \cos(\lambda)$$

$\phi \rightarrow$  Angle between vector to glide plane &  $\sigma_1$

$\lambda \rightarrow$  Angle between glide direction and  $\sigma_1$

Max is at  $45^\circ$  because  $(\cos 45^\circ)^2 = 0.5$



## Critical Resolved shear stress (CRSS)

- Component of shear slip resolved in the direction of slip, necessary to initiate slip

$$\tau_{RSS} = \sigma_n = \sigma (\cos \phi \cos \lambda)$$

← Schmid factor

Schmid good for single crystals. For polycrystals

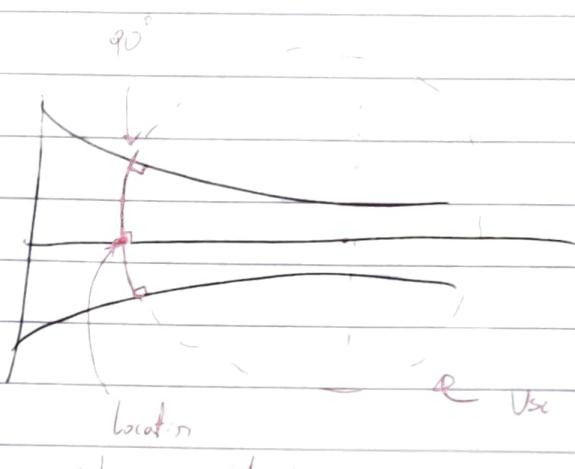
Mention of here is tensile, perhaps engineering convention!

Taylor G.I.  $\rightarrow$  5 slip sys

Slip factor as proxy for orientation

Hm. Okay

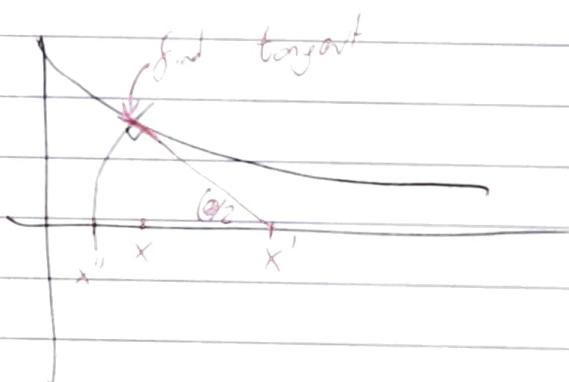
As  $\theta$  increases so does  $\Omega$



Solid angle  $\rightarrow \Omega$

Work:

1. Describe the sample shape  
as  $y = f(x)$



2. Find tangent equation (differentiate) at  $x$

3. Find intercept with  $x$ -axis ( $x'$ )

4. Find length of this line (sphere rad.  $\rightarrow$ )

$$5. x'' = x' - (r)$$

6. Find angle with  $x$ -axis ( $\theta/2$ )

Area of sphere =  $A = \frac{4}{3}\pi r^2$

7. Find the area subtended  $\rightarrow$  Solid angle  $\Omega$

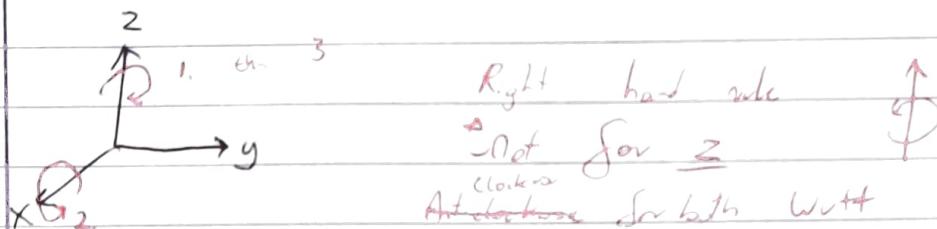
To find cone cap area:

$$\frac{\Omega}{4\pi} \times A \quad \left\{ \frac{\Omega}{4\pi} \times \cancel{\pi r^2} = \Omega r^2 \right.$$

Look through Euler angle diagrams

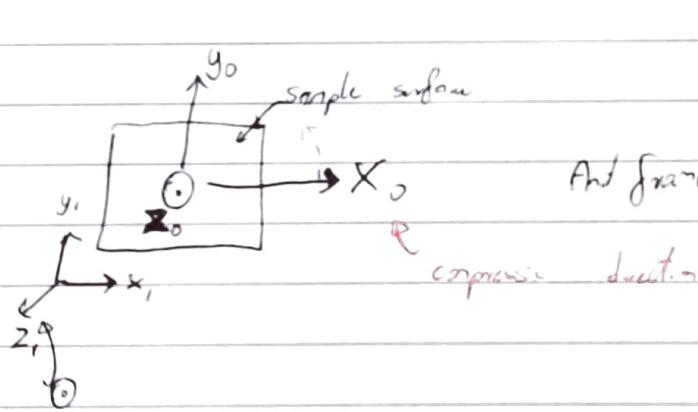
Three Euler angles  ~~$\phi_1 \phi_2 \phi_3$~~   $\phi_1 \phi_2 \phi_3$

- 1) Rotation of  $\phi_1$  about the  $z$ -axis
- 2) Rotation of  $\phi_2$  about the rotated  $x$ -axis
- 3) Rotation of  $\phi_3$  about the rotated  $z$ -axis

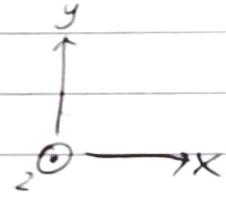


Plot upper hemisphere

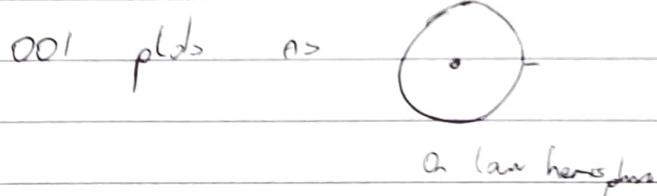
Relation to Sample:



Ant frame is also



long-axis parallel to  $x$



in reference frame,  $x$ -axis is compressive direction

Ax 3-New  $\rightarrow$  c-axis

c-axis vertical

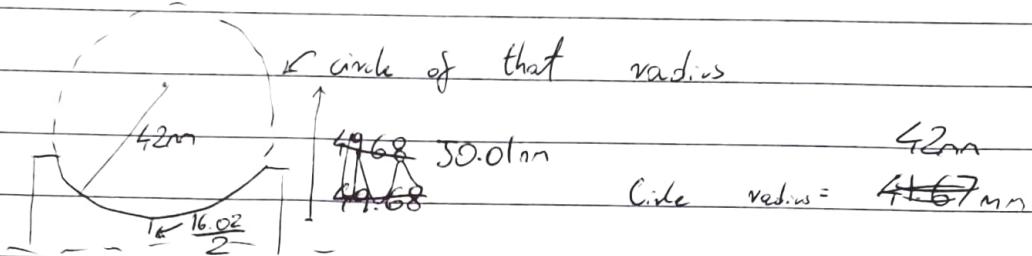
a-axis north

In the end,  $\frac{\text{load}}{\text{max stress} \times \text{area}} = \text{the pressure}$

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For Geometry

13 APR 2021

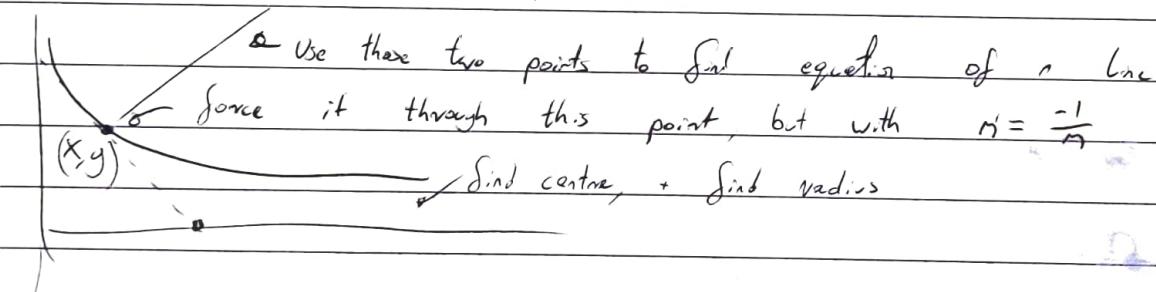


$$(x-a)^2 + (y-b)^2 = r^2$$

$$y = \sqrt{r^2 - (x-a)^2} + b$$

$$\text{MPa} \cdot \text{m}^2 \quad (10^6) \text{ Pa} \quad (10^{-6}) \text{ m}^2$$

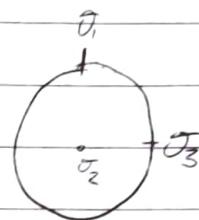
Circle center (50.01)



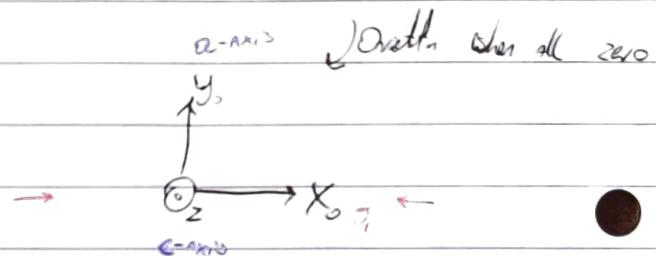
Clark Cone opening angle theta



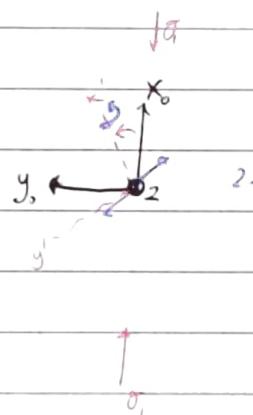
I want a stenoc that looks like:



My axes are such that



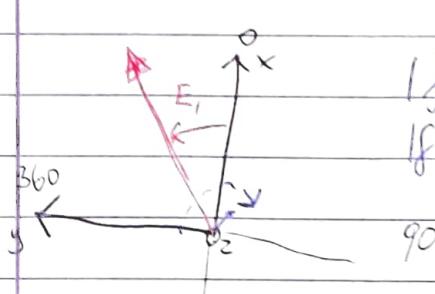
So let's say my coordinates are such that



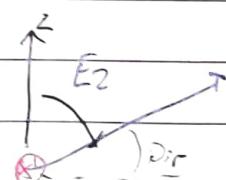
Upper hemisphere, all rotations +ve

Plunge: of our  $\theta$ , add  $180^\circ$  to  
Azimuth

For Azimuth



Azimuth  
If  $0-90$ , Strike is  $+90^\circ$  from strike  
If  $90-180$ , Azimuth is  $+270^\circ$  from strike



If less than  $90^\circ$   
 $D_p = 90 - E_2$   
if more than  $90^\circ$   
 $D_p = E_2 - 90$

Feed in some points for check:

0, 90 should be ~~090~~

Good 1:1 check

as there were errors

1, 89 should be 089, ~~091~~

(i. the calc  
I think my taking  
is alright)

0, 40 should be 090, 50 ✓

95, 30 should be 355, 60 ✓

95, 110 should be 175, 20 ✓

110, 110 should be

Select best image for analysis:

Chaneling nodule shows the clearest twins  
 ↳ Slightly oversized.

Ennis's Dec 10 email:

Use Euler to work out 3 possible orientations  
 plot that on image, and see which scratches  $\rightarrow$  fine?

Work at Euler. Render vector will be  $\uparrow$  to e-plane

But actually if I plot 3D line on 2D image, it  
 should project just right  $\rightarrow$  Or actually plot the plane  
 for  $z = 0$  to  $\approx 1$  + small range parallel to look at  
 But best is 1 first intersect

Jobs for Tuesday? + Place figure in matlab and  
 coordinate with  $(0, 0)$  as origin.

Right-hand side is a bit off-axis, so given coordinate  
 will not work perfectly

Direction comes from Plunge / Plunge Azimuth in right-handed system:

$\theta$  - Azimuth

$\delta$  - Plunge

$$\cos \delta \sin \theta = \cos \alpha$$

$$\cos \delta \sin \theta = \cos \beta$$

$$-\sin \delta = \cos \gamma$$

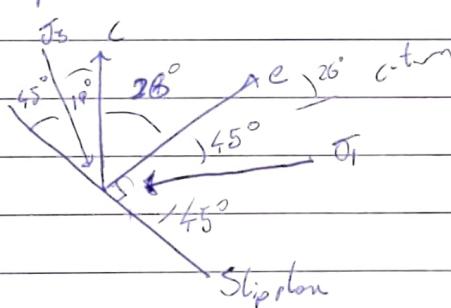
t

Because I'm using upper hemisphere, I will probably want a neg. sign somewhere in this equation

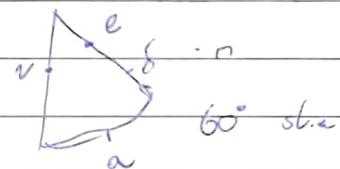
Success! All angles but for Grain 26 are about  $58^\circ$ , as expected

$\ell_{80^\circ}$

C-E plane



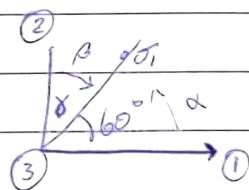
They'll  $\rightarrow$  no  $\alpha$ -axis, but probably a  $\gamma$ -axis no.  $\beta$ -axis through



The plane containing  $a$  is 30 degrees from nearest  $\alpha$ -axis and 60° from nearest  $\gamma$ -axis

But, it also is  $90^\circ$  from the next  $\alpha$ -axis

Eric's Diagram:



Current thought process  $\rightarrow$

Find direction cosine from 3 Euler angles  $\rightarrow$

- Find rotation matrix of one system onto the other
- Establish all angles in system 1
- Carry out the rotation

If no success see what Enric says

Okay. Using even Euler angles of Host, should be able to guess directions of ~~host~~, and hence also or. twins

Can use existing grains as a benchmark here.

Good idea  $\rightarrow$  Application will be non-trivial  $\rightarrow$

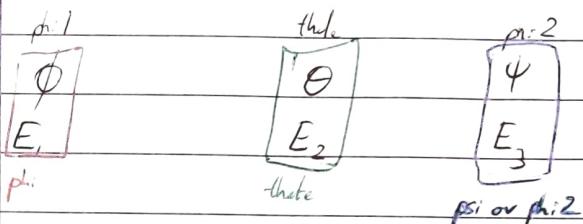
Why does Enric use plane  $60^\circ$  from  $\text{D}_\text{host}$ -axis?  $\rightarrow$  m-axis?

Doesn't e lay a m-axis?

First rotation:  $\phi$ : 1 range -180 to 180 about z

Second rotation:  $\theta$ : range 0 to 180 about x

Third rotation:  $\psi$ : 2 range -180 to 180 about z



$$R = \begin{bmatrix} a_{11} & a_{12} & a_{13} \\ a_{21} & a_{22} & a_{23} \\ a_{31} & a_{32} & a_{33} \end{bmatrix}$$

$$a_{11} = \cos \psi \cos \phi - \cos \theta \sin \phi \sin \psi \quad E_3$$

$$a_{12} = \cos \psi \sin \phi + \cos \theta \cos \phi \cdot \sin \psi \quad E_2$$

$$a_{13} = \sin \psi \sin \theta \quad E_2$$

$$a_{21} = -\sin \psi \cos \phi - \cos \theta \sin \phi \cos \psi$$

$$a_{22} = -\sin \psi \sin \phi + \cos \theta \cos \phi \cos \psi$$

$$a_{23} = \cos \psi \sin \theta$$

$$a_{31} = \sin \theta \sin \phi$$

$$a_{32} = -\sin \theta \cos \phi$$

$$a_{33} = \cos \theta$$

Do it and do trial and error

Input:

$$\begin{matrix} & \text{Expected result?} & \checkmark \\ \begin{matrix} v_1 \\ v_2 \\ v_3 \end{matrix} & \begin{matrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 1 \end{matrix} & \end{matrix}$$

z-axis points towards x ✓

x-axis points towards y ✓

z-axis points towards z ✓

90, 0, 0

✓

Success.

Figured out how Eric's 'corrections' work for stereonets but I don't think I need them. → actually removes potential symmetry errors

If I do all three Euler rotation cosines:

$$\begin{bmatrix} A_{11} & A_{12} & A_{13} \\ A_{21} & A_{22} & A_{23} \\ A_{31} & A_{32} & A_{33} \end{bmatrix} \quad \begin{matrix} n\text{-axis} \perp \text{to } c \\ a\text{-axis} \perp \text{to } c \\ c\text{-axis} \end{matrix}$$

Look up Rodrigues rotation formula

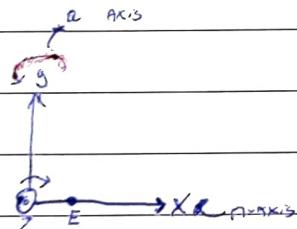
$$\underline{v}_{\text{rot}} = \underline{v} \cos \theta + (\underline{k} \times \underline{v}) \sin \theta + \underline{k}(\underline{k} \cdot \underline{v})(1 - \cos \theta)$$

→ uses right hand rule

$\underline{k}$  is the unit vector describing rotation axis

$\underline{v}$  is the vector that will rotate

$\theta$  is the angle of rotation



So,  $c$ -axis, or  $\underline{z}$  is the  $\underline{v}$

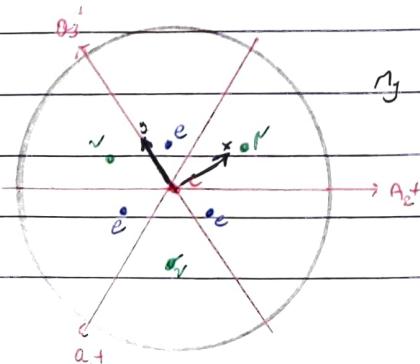
$a$ -axis, or  $\underline{y}$  is the  $\underline{k}$

Going from  $\underline{z}$  to  $\underline{a}$  is in positive sense.

(However machine conventions may be a problem?)

Going  $26^\circ$  gets me to  $\underline{a}$ . Actually, at  $+\alpha$ , this gets me to  $\underline{a}$  in  $\underline{v}$  plane.

\* Will probably get me to a point  $60^\circ$  off the  $a$ -axis, so better  
Either way, solution can be shifted about. Add  $60^\circ$  to go from  $\underline{v}$  to  $\underline{a}$   
catering  $n$ -plane  $\rightarrow$  around  $z$ -axis of course



$v$  new ht  $\underline{x}$

Apply it! → by finding two angles  
first to confirm:

1. Rotate around  $\underline{y}$  by  $52^\circ$
2. Rotate around  $\underline{z}$  by  $60^\circ$  if necessary

But first get the angles to deviation cosines...  
Success!

Confirmed that it is possible to rotate the measured host into an orientation that corresponds with the measured twin.

This involves, as previously stated:

1. Rotate by  $52^\circ$  around  $y(\alpha)$  axis so that  $z + z''$
2. Rotate  $z'$  by  $60^\circ$  around  $z$  so that it hits the  $c-e$  plane
3. Further rotate by  $0^\circ, 120^\circ$ , or  $240^\circ$  to align with 1 of 3 symmetries.

Usually  $z''$  is within  $2^\circ$  (sometimes up to  $10^\circ$ ) from measured value.

Next task: Convert direction cosines to zenith/phase so they can be put on stereonet.

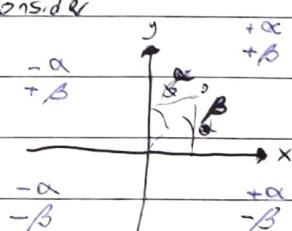
Potential equations:

$$\theta = \arctan(\cos \alpha / \cos \beta)$$

$$\delta = \arcsin(-\cos \theta) \quad \leftarrow \text{One again consider negative for upper hemisphere.}$$

Azimuth	$\cos \alpha$	$\cos \beta$	$\theta$		Or even -1/2/1 to deal with symmetries	Actually may need to change signs of ALL direction cosines
$0-90$	+	+	$\theta$	✓		
$90-180$	+	-	$180 + \theta$	?		
$180-270$	-	-	$180 - \theta$	?	Questionable	
$270-360$	-	+	$360 + \theta$	?	All correct ✓	

Consider:



Success!!

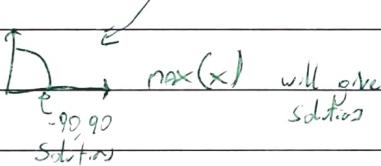
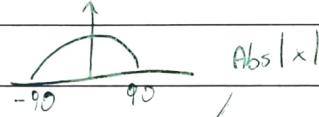
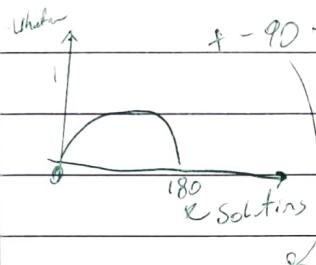
Cosine-to-Azimuth works well, and in a predictable fashion.

Next: Plot  $\sigma_1$ -c-e plane, and  $\sigma_3$  for known gneiss

For Sigma Sindy

I need to select when  $\Delta\theta$  is close to  $0^\circ$  or  $180^\circ$

To find value closest to  
 $0$  or  $180$



Actually think about implications.  
Is it better to rotate it around?  
Test this  
Grans: 90, 163, 292, 311

In this case:

g-axis always  
Rotate by  $180^\circ$  around y? Is it necessary?

See what starcode does

Grans: 135 is problematic  $\rightarrow$   
because  $16^\circ$  bottom estimate  $\rightarrow$  intol  
two c-axis

Maybe write a function that will write tables with  
Headers otherwise the text files may get  
confusing

Tomorrow try to plot extension direction

Proposed method for plotting traces  
of  $e$ -planes and  $v$ -planes

Once pole to a plane is found,  $\hat{p}$ , we want the trace of the plane it represents, as it intersects the  $x-y$  plane.

This trace is perpendicular to the pole  $\hat{p}$ .

This plane is also always perpendicular to the  $z$ -axis.

$\curvearrowleft$  in geographic reference  
(0, 0, 0) ECU

Hence, do cross-product  $\hat{p} \times \hat{z}$  to get the direction of the trace.  $\rightarrow$  which should have only  $(x, y)$  components

Because  $\hat{z}$  is not always perpendicular to  $\hat{z}$ , output is not a unit vector, and will need to be normalised.

Then do necessary adjustment (lengthen vector) so it shows up well when overlaid on the image

Should work. Task set next day

- Reuse lots of signal code probably

$\rightarrow$  + first ~~to~~ test on known grain

Poles to cleavage planes are  $44^\circ$  from c



Establish E-plane  
and R-plane  
poles for all grain

(cleavage plane) May be confirm it with twin plane  
63.9 (or  $26.1^\circ$ ) according to 2016 paper

Note from Miller-Bravais indices:  $\{0001\}$  are directions of poles to planes  
 $\{0001\}$  are planes  
 $\{0001\}$  are families of directions ✓  
 $\{0001\}$  are families of planes

Spies 1980  $\rightarrow 44.5^\circ$  for c-r

Emin confirms  $44^\circ$  in email

Try to confirm this using:

$$\cos \phi =$$

Actually dead end ✓

Try inputting indices as  $[h k l, \frac{3}{2}(\alpha_c)^2 l]$ , which should be a normal.

$$\alpha = 10$$

$$c = 8.5$$

$$[h k l]$$

factor = 0.922 or perhaps 1/

$$i = - (h + f)$$

↑ unnecessary

Directions represented by  $u v t w$

need all 4

e is  $(\bar{1}018) <40\bar{4}1>$

r is  $(10\bar{1}4) <\bar{2}021>$

c is  $[0001]$

f is  $[\bar{1}012] <2\bar{2}01>$

Plotting of traces figured out - now just worry about mapping results to image

Grains 8 and 198 are the least associated ( $6 \pm 8^\circ$  from geographic z-axis respectively).

Everything else seems to work fine

For some grains, like 19 or 53, the  $c_1/c_2$  traces don't really line up - maybe  $10^\circ$  mismatch. The closest solution however is the one picked up by the EBSD

Change of tactic: instead of trying to plot the traces on top of the image in Matlab, try to do it in illustrator. Then are some advantages to this:

1. No axis; no need to worry about
2. No grain coordinates to worry about
3. Easier to adjust size as needed
4. Easier to shift about as needed

→ However add <sup>Grain</sup> number to axes so it is clear what I am talking about

→ I AM AN IDIOT!

- Everything is fine. I didn't set up the chart dimensions right, so they came out a bit "squashed". After rotation everything is okay. x/y lin [-1 1] phasep [1 11]

Anyways, proceed with this

Success! Generated 322 images (for each grain)

Need to overlay them on chemoly images now, and collect twins  
in a separate spreadsheet Solutions

Grains that can't really be analysed:

6 - v overlaps the only e(3)

F-slip analysis successful  
from grain 16

7 - Grain doesn't really exist on chemoly

10 - Two v-axes place on grain e(2) - exactly

15 - No twins, but it's small and fractured so inconclusive

18 - e(2), but small nss-match  $\approx 10^\circ$

22 - e(2) possibly overlapped by (v)

Δ

Created a word document for these, probably a better route

204 - perfect grain !!

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Good v-image matches solutions

17 & 19 are the same grain  $\rightarrow$  good example of ~~the~~ our limitations.

8 Convert from Direction cosines to Euler angles.

- Input as will make using older code 10x easier

$$a_{31} = \sin \theta_2 \quad \text{E}_2$$

$$a_{32} \cos \theta_1 = -a_{33} \sin \theta_1$$

$$\frac{a_{32}}{-a_{33}} = \frac{\sin \theta_1}{\cos \theta_1} = \tan \theta_1 \quad \text{Here: } \theta_1 = \arctan \left( \frac{a_{32}}{-a_{33}} \right)$$

$E_1 \ E_2 \ E_3$   
 $\phi \ \theta \ \psi$

Notably that this is a different convention. May come back to it.

Worry about +/- solutions? later. Try from 1st principles:

$$a_{33} = \cos \theta \quad \theta = \arccos(a_{33}) \quad E_2 \text{ solved} \checkmark \theta \text{ known}$$

$$a_{31} = \sin \theta \sin \phi \quad \arcsin \left( \frac{a_{31}}{\sin \theta} \right) = \phi \quad E_1 \text{ solved} \checkmark \phi \text{ known}$$

$$a_{13} = \sin \psi \sin \theta \quad \arcsin \left( \frac{a_{13}}{\sin \theta} \right) = \psi \quad E_3 \text{ solved} \checkmark \psi \text{ known}$$

Doesn't quite work. Unsurprisingly

Problematic when  $E_1$  is over  $90^\circ$

Better Soln: Someone already wrote good (except) code for this that takes account of all orientations and zeros ( $1/\theta \rightarrow \infty$  problem)

Adapted that and it works perfectly!

Object.:

Write code that will "remove" the twins from ANY file, provided grain No. is in the first column.

What  $\rightarrow$  means is: Removes problematic grain rows, using problematic grain "key".

This will make the tables of data less troublesome when feeding into other code.

Complete:

Now try to get `signal` and `Sign3` into stereonet

Note: "Extract-Convert" software will do rotations to `Signal` and `Sign3` when nicely adjusted.

Input run "Extract convert" and "Prepare for Stereonet" together.

$\leftarrow$  To generate poles/bedns

$\leftarrow$  To remove problematic and prepare for plotting

If need stereonet in Core Rtk "Extract convert" to generate Eulers. Then run "Remove\_Problematic".

The user part of  $\left\{ \begin{array}{l} \text{"Prepare for stereonet"} \\ \text{"Remove_Problematic"} \end{array} \right\}$  to summarise if needed  
 Perhaps add this to "Extract convert"  $\leftarrow$  perhaps best not to do it yet, as it will be easier to append pressure, densities etc to "intacted" data - have row 1 = Grain 1

NO  $\rightarrow$  will mess up this part

Catch up on work  $\rightarrow$

Inverse pole figure first.

Start by doing a inverse-order negative sense Euler rotations.

(That will tell you how to get the crystal back to geographic axes. In some terms, x-axis is the sigma!)

See where it gets data. Rotate by  $90^\circ$  around  $[001]$  & crystallographic if needed (if statement)

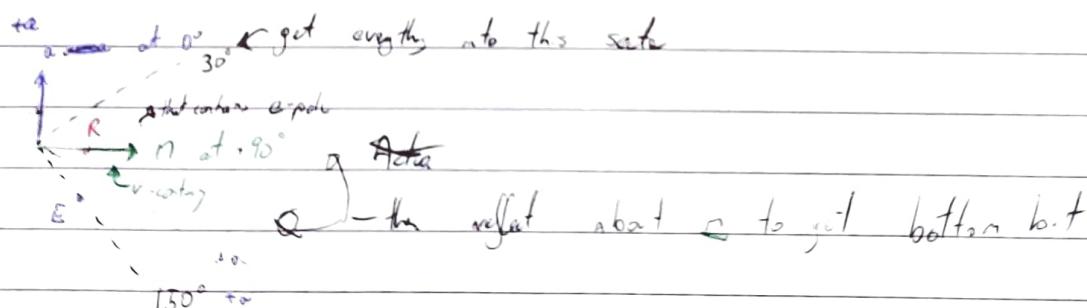
Try this. Then try this  $\xrightarrow{\text{from Euler}}$  for apoles. See where it goes.

odd hours pole Eul to calc Lat

Seems to work. Nice!

Split up based on twin law and remove problematic

As it stands:



Problem  $\rightarrow$  If A is  $\rightarrow$ , I might have rotated in RF by  $\omega$

That about this again for me  
as it is wrong

For Schmid factor:

Do this for a list of non-planar E-planes

$$\eta = \cos(\phi) \cos(\lambda)$$

$\phi$  + Angle between e-pole and  $\sigma_1$  Always  
 from [100] & x-parallel  
 Extract  $\approx 26^\circ$  on CE plane

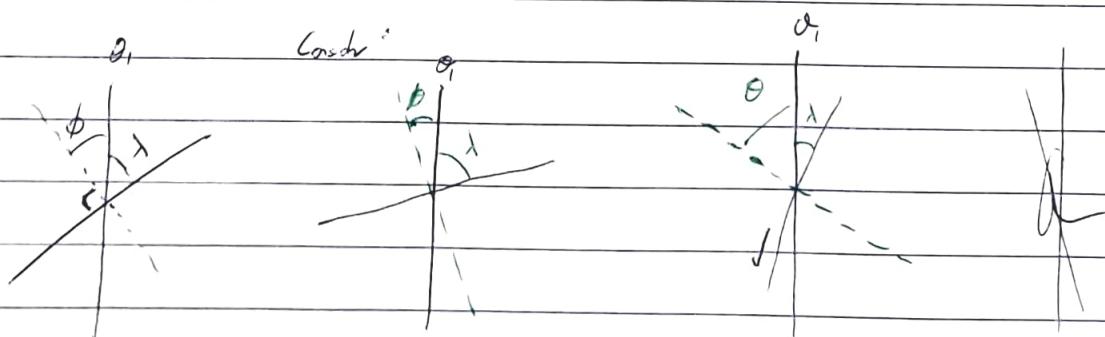
$\lambda$  + Angle between Slip vector

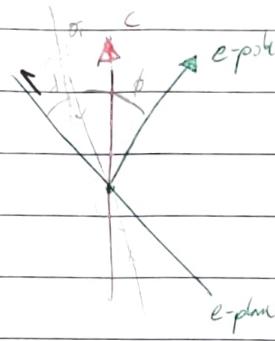
$-64^\circ$   
Along CE plane

Only 64 tics with positive Schmid factor

↑ And they also plot right in the middle of the stereonet.  
Probably got exactly the wrong ones

Actually, by definition, maybe negative are in fact next to be  
Solutions:





I think  $\lambda$  should be the ~~angle~~  $180^\circ$  - angle between vectors,  
as that then finds the actual  
angle to the slip vector

Now 182 good twin

$F_p$  poles: 150,

$F_x$  great circles to do Spars twin - zone plot.  
Either  $\lambda$  or  $\delta$  not  $= 90^\circ$  for zero resolved stress

So poles over e-poles  $\Rightarrow$  slip vectors

e-poles:

150, 26

30, 26

270, 26 + Rember upper hemisphere, & it work great

1st: 330, 64

210, 64

90, 64

Add to problematic list:

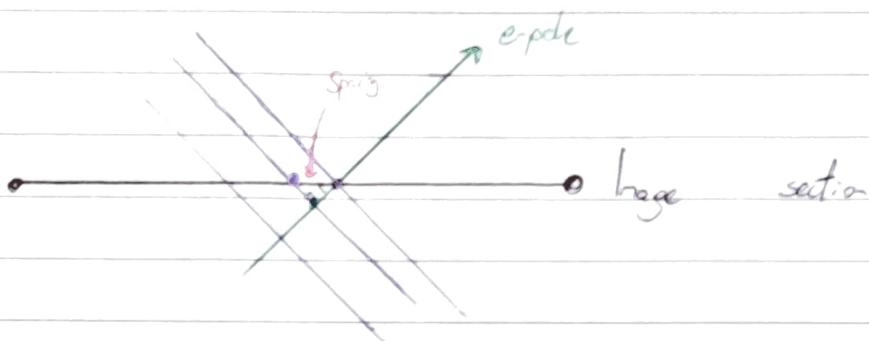
181 - Actual solutions are 3 and 2 - I must have made a mistake earlier

283 - Same story  $\rightarrow$  Actual solutions are 2 and 3

Carefully checked that  $\rightarrow$  f will need to generate figures

Again  $\rightarrow$  Anch checked if School factor is still good.

Find the true density. <sup>Corrected for image section effects</sup>



$$\text{Density} = \frac{\text{Number of tanks}}{\text{Span length}} \quad \text{in tanks}$$

$$\text{Span} = 1/\text{Dist} \quad \text{in m/tank}$$

$$\cos \theta = \frac{\text{adj}}{\text{hyp}}$$

Span behavior of dist

$$\cos \theta = \frac{\text{corrected span}}{\text{apparent span}}$$

$$(\text{apparent span})(\cos \theta) = (\text{corrected span})$$

$$\left( \frac{1}{\text{apparent dist}} \right) (\cos \theta) = \frac{1}{\text{corrected dist}}$$

$$\frac{\text{apparent dist}}{\cos \theta} = \text{corrected dist}$$

• Goal - find 1/p of e-pole for all sections now

All done ✓

## Tasks for today:

- Extract a list of good grains + twins + diocles + pressure from the ~~measure~~ spreadsheet.

- Sort them into ~3 size categories.

↳ Maybe more - will see

[0-50, 50-100, All above 100]

• Plot twin activity against Schmid factor

• Twin activity against Pressure

↳ Not very fruitful

↳ Not very fruitful

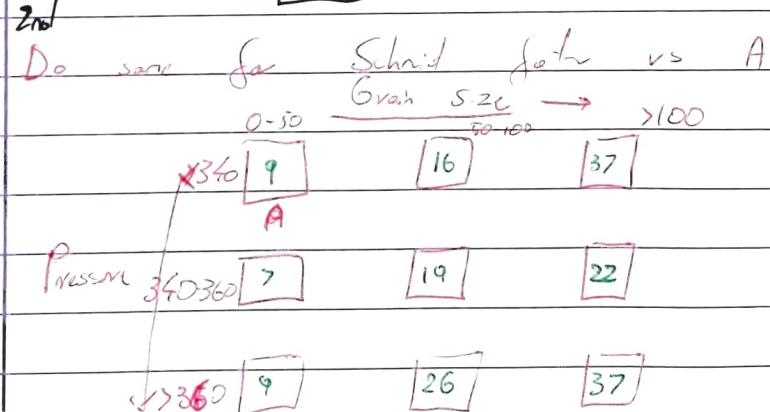
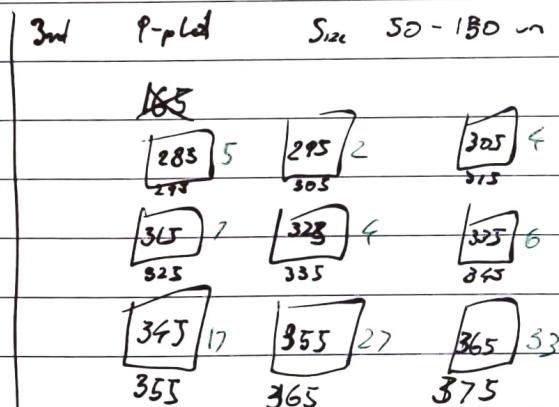
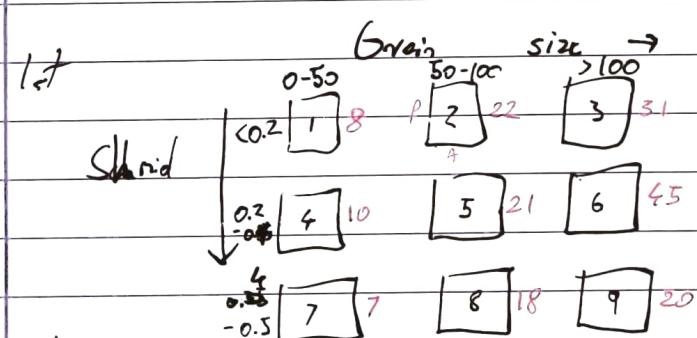
Then further divide into groups by Schmid factor and plot:

- Twin activity against Pressure for each Schmid group

## Matlab storage Game!

10 MAY 2021

1	2	3	4	5
Grain	Schmid	Twin Activity	Diacl	Dif pressure



Gradient of  $\frac{\text{Shrid}}{\text{Activity}}$  may be related to pressure

→ Smaller pressure = Bigger gradient

Try plotting Activity/Shrid vs Pressure → Should see a positive relationship

Still try this in grain-size graphs

Thesis layout:

## Introduction

- Explain the term polycrystallinity
- Explain Cubic (briefly) and how it makes marble rocks - incl. Common Marble. + Show some "habit" diagrams. "Cubicographic projections"
- Describe twinning - both growth and deformation
- Explain the basis of "Experimental calibration".
- Briefly introduce my approach - how thanks to EBSD I will now be able to use orientations. And how current methods are 30 MPa limited.

## Background:

- Work by Ernie in 80s/90s - all the dependancies
- Inversion in lower crust
- Waves from 30s
- Sprays
- Karakoram from 07 + other potential applications.

## Experimental set-up and data collection

- Drawing, numbers, etc.  $\rightarrow$  Starting processes, time
- Dog bone shape
- Sample cut in half and EBSD'd by Dave Wallis

## Data generated:

- EBSD Map, + Orientation data (Explains EBSD + reference)
  - Channeling image
  - Band contrast image
- For habits and some twins  $\rightarrow$  Euler Angles + Bragg notation
- e ①  
= Grain Area

## Theory: Orientation Theory

- Start with Objective: To use  orientation information
- Basically (Part 1)

## Grain size and pressure Theory:

- How grain size is worked out (Fast diameter)
- How pressure is worked out (Part 2)

## Analysis (Orientation)

- Start by analysing orientation of c-axes, E-poles and angles between them. (Show how Euler  $\rightarrow$  Casini in Bunge convention)
- How other orientations are worked out. Mention Rodriguez rotation, how  $CE$ -axis is  $60^\circ$  off  $CR$ -axis, and how there are three solutions.
- Turner Weiss on known grains. Mention problematic 135 + 26.
  - This is a confirmation, but show  $30^\circ$  offset - comment how this may be a pre-existing fabric
- THEN how 3 solutions for each were generated. (as e-polys)
  - These e-polys were transformed into traces by x-product with sample z-axis. → This gives traces. Give some examples.

Those are overlaid on image and solutions recorded.  
 Show some perfect and some commonly problematic grains. Mention stats. How many of the 323 grains are good, how many poles, etc.

- Do Sch. Turner-Weiss analysis with these  $\rightarrow$  This will still be problematic cause of negative Schmid factor.
- Do Schmid factor corrections  $\rightarrow$  Then Turner Weiss again
- Explain how inverse pole figures are generated, and do one for (0, 1, 2, 3 twin zones) as Spiers
- Counting twins and e-normal corrections

~~Grain size and pressure thys has~~

- Grain size as fast diameter

Pressure by spherical shell approx, as the stress at sample surface must = 0

- Show some pressure - distance graphs.

## LINKING PRESSURE WITH TWINNING

- Show case the graphs and explore potential correlations.

- Explain why it's important to split into grain size groups.

↳ Explore splitting into Schmid factor groups

↳ Add the one last residual shear stress b.t if the peak

Rescale Pressure to whatever it is Eric called it

### Discussion:

What's in the word file

A lot more twin planes collected

Discuss some core options:

- Bigger source is grain size - problem further studies with lack of distinct grain size distributions

- Potentially some planes identified as twins are NOT twins but slip on non- $\bar{c}$ , non- $\bar{a}$  planes that overlaps. R

Discuss Neighbor effect

Also pre-existing

- Suggest further manipulation of the data

### Conclusions:

- Twin Weiss Analysis works great

- Zones predicted by Spivars generally seen

- A LOT more twin planes available to study thanks to tree author

- Factor analysis of transformed data (as in Fabio's email)

Acknowledgments: like in Word file

WORK:

①

## Presentation plan

Intro

→ Define terms

→ Why it's useful

→ Calcite geometry

→ Data available

→ Why calcite is useful

## - Define Palaeopiezometry ①

1. Introduce the concept of <sup>→ Snapshot of analysis</sup> recording pressure at depth

- Like with metamorphic paragenesis analysis.

2. Defn. differential stress. \*

3. Explain the ut. of a ~~piezometer~~ & tell stress state in Valley, but also study of Regional tectonics  
→ in 2007 a shear zone was defined thanks to Calcite palaeopiezometry

4. Intro into calcite → Start very basic.

→ Nature of calcite Carbonate → Main constituent of rocks

5. Calcite structure →

Rhombohedron

- What's important is that it has three a-axes,

and optional c-axis.

All a-axes are identical, but one of different length to c-axis & this varies with temperature  
thanks Pal & Steve.

Going back, the important is symmetry.

There is reflection and inversion symmetry in Calcite too, but all we care about is rotation, as all our action happens <sup>right</sup> on the mirror-planes. Stereonet Unto that it's upper hemisphere6. Step back to twinning - Definition (reference)7. ~~We regular twinning can happen on three~~ Twining we are interested in is called the c-plane twinning, because it occurs on c-planes↓  
Point it out- E<sub>pol</sub>

(2)

8. Show where a plane is on calc. But notice this isn't right. What is important is that it lies on a mirror-plane, ~~so~~ lies 26 degrees away from the c-axis and there are 3 of them.

9. Show how to switch topic to shear stress  
Show the equation of schmid but don't call it that yet

Explain what the terms mean in 2D

Show a plot of  $\cos\theta \times \sin\theta$  to show a max of 0.5 when  $\theta = 45^\circ$ .

I.F. Resolved shear stress largest when stress vector is at  $45^\circ$ .

→ Called "Schmid Factor"

10. Now sum it all up: e.g. And applies in 3D as well  
Show angular relation

- Sigma 1 at  $71^\circ, 26 + 45^\circ$  from Host, toward E.pdc.

- Note that in calc. there can be more than one active slip system - Question

This forms a basis of one analysis performed. If we assume all the terms are optimally oriented, and sigma 1 is at  $45^\circ$  from the twin plane, and we plot the sigma 1 estimates for multiple grains, we shall know where the actual sigma 1 is.

Twinning - Weissen

11. This is what the graph shows. Of course almost none of the grains will be perfectly oriented, hence we have a spread. But after contouring, we see that the prediction is spot-on. North and South poles are the actual points where set stress directions

(3)

Channely image fragment with traces

27 MAY 2021

12. ~~But how did~~ But that's part of data analysis.  
First we had to gather all the data.

I didn't have the time to ~~present~~ showcase all the data I was presented with, but let me show you the most essential parts.

[Channely image - somehow in the corner]

This is ~~another~~ an image which shows ~~a~~ several slip systems traces of many ~~slip~~ glide slip systems. Some of them will be active, and some will not.

• We also know the orientation of each host grain.

This isn't shown on the image - it's part of general data generated by the EBSD

13. Traces appear Remember, the geometry we learned about before, through some complex math we can predict ~~where~~ the traces will fall - what the expected trace of each plane should be. This is shown as the coloured lines on the graph.  
Essentially, by analysing this image, which contains over 300 grains, but we only see a handful on the screen now

14. Table with no. of twins collected

This method allows for collection of 216 twin planes but after filtering of problematic data this number dropped to 152 [Ask why so many grains were rejected.]

Filter-off

This is a ~~~~~ 200% apparent or ~~what the~~ the volume of data that EBSD has collected

[List of these]

⑦

15: To finish off, few details why we might want to look into calcite further:

- Research has shown that it is Temperature insensitve  
 ↗ Pressure ins. Stress-rate insensitve  
Good thing

~~Here can tell pressure an ~~Grain-size sensitive~~~~

↳ needs more work

16. Qs

Stress magnitude graph → I think very new data, but it requires more work. I think ~~the~~ ~~so~~ I may have more success if I re-evaluating ~~as~~ ~~as~~ how I define and measure twinning ~~at~~ today.

But that's all - we have time for today. Any Qs?

17. Qs?

18. Refun