

# Evaluation of elastic modulus of $\text{Li}_2\text{S-P}_2\text{S}_5$ glassy solid electrolyte by ultrasonic sound velocity measurement and compression test

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Elastic modulus is an important factor of solid electrolytes for an all-solid-state battery as a next-generation battery. In this study, Young's moduli of dense pellets of the  $\text{Li}_2\text{S-P}_2\text{S}_5$  glass solid electrolytes prepared by room temperature pressing and hot pressing were investigated. The Young's moduli of  $\text{Li}_2\text{S-P}_2\text{S}_5$  hot-pressed pellets measured by ultrasonic sound velocity measurements were 18–25 GPa and those of cold-pressed pellets were about 14–17 GPa. The compression test was also done to determine Young's modulus. The Young's modulus of  $\text{Li}_2\text{S-P}_2\text{S}_5$  glasses increased with increasing the  $\text{Li}_2\text{S}$  content in both hot press and cold press pellets. The Young's moduli were lower than those of oxide based solid electrolytes.

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## 1. Introduction

There is an increasing demand for large-scale secondary batteries with high safety and energy density for the application to power sources for electric vehicles and energy storage systems for renewable energy.<sup>1)</sup> All-solid-state lithium secondary batteries using inorganic solid electrolytes are a potential candidate for next generation batteries with extremely high energy density. Conventional lithium-ion secondary batteries have been widely used as a high energy density battery. However, they contain liquid electrolytes using flammable organic solvents and thus have safety concerns. Improvement of the safety of batteries is expected in all-solid-state batteries by using inorganic solid electrolytes instead of organic liquid electrolytes.<sup>2)</sup> The improvement of energy density is also expected in the all-solid-state batteries because high capacity electrode materials such as elemental sulfur and lithium sulfide will be used.<sup>2)</sup>

Many challenges should be done for realizing all-solid-state batteries. Mechanical properties of solid electrolytes are important for being developed. For example, solid electrolytes should have high processability to construct intimate solid-solid contacts between electrode active materials and solid electrolytes.<sup>3)–5)</sup> The intimate contacts between solid particles are essential to achieve high performance such as high capacity, long cycle life, and high rate capability. Both electrode active materials and electrolytes are solid in the all-solid-state batteries; it is thus difficult to construct the intimate contact between electrode and electrolyte compared to the conventional batteries using liquid electrolytes. Furthermore, the contact should be held during charging and

discharging although volume of the electrode active materials changes during charging and discharging. The mechanical properties of electrode and solid electrolytes should affect the electrode/electrolyte contacts during cycling.

Elastic modulus is an important mechanical property to investigate the electrode/electrolyte contact. The understanding of elastic modulus of the solid electrolytes is necessary for the development of new solid electrolytes with suitable mechanical properties for all-solid-state lithium secondary batteries. Some papers report the elastic modulus of lithium-ion conducting oxide solid electrolytes.<sup>6),7)</sup> However, there are few papers which reports elastic modulus of highly conductive sulfide solid electrolytes because of the difficulty of the experiments in air atmosphere.

Sulfide solid electrolytes as the  $\text{Li}_2\text{S-P}_2\text{S}_5$  glasses show lithium-ion high conductivity.<sup>8)–12)</sup> The  $\text{Li}_2\text{S-P}_2\text{S}_5$  glasses are potential candidates for solid electrolytes of all-solid-state lithium secondary batteries.<sup>3),4),13)–15)</sup> Recently, we achieved the preparation of almost fully dense pellets of the  $\text{Li}_2\text{S-P}_2\text{S}_5$  glasses by hot press from the glass powder at around their glass transition temperatures.<sup>16)</sup>

The mechanical properties like Young's modulus as well as electrical properties were varied with the composition in the  $\text{Li}_2\text{S-P}_2\text{S}_5$  glasses. It is important to clarify the relationship between composition and the properties. Furthermore, the electrolytes were used as powder compressed pellets in the bulk-type (powder-type) all-solid-state batteries. Therefore, the investigation of mechanical properties of the compressed pellets as well as fully dense pellets is valuable.

In this paper, the dense pellets of the  $\text{Li}_2\text{S-P}_2\text{S}_5$  system glassy solid electrolytes were prepared by room temperature pressing and hot pressing. Young's moduli of the prepared dense pellets were investigated by ultrasonic sound velocity measurements and compression tests. The relationship among Young's modulus of the pellets, the glass composition, applied pressures, and temper-

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**Table 1.** Velocities of longitudinal and shear waves, density, shear modulus, Young's modulus, Poisson's ratio of  $x\text{Li}_2\text{S} \cdot (100 - x)\text{P}_2\text{S}_5$  (mol %) solid electrolytes prepared using various molding pressures and temperatures. The shear modulus, Young's modulus, and Poisson's ratio were calculated using an ultrasonic sound velocity technique. The  $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$  ceramic pellet was prepared by sintering at 1230°C, and the  $60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$  glass pellet was prepared by the melt quenching method

Sample	Molding temperature /°C	Molding pressure /MPa	Velocity of longitudinal wave $V_L/\text{m s}^{-1}$	Velocity of shear wave $V_S/\text{m s}^{-1}$	Density $\rho/\text{g cm}^{-3}$	Shear modulus $G/\text{GPa}$	Young's modulus $E/\text{GPa}$	Poisson's ratio $\nu$
$\text{Si}_3\text{N}_4$	—	—	10,830	6,050	3.1	114	289	0.27
$\text{Si}_3\text{N}_4$ (in polymer pack)	—	—	10,770	6,060	3.1	114	289	0.27
50 $\text{Li}_2\text{S} \cdot 50\text{P}_2\text{S}_5$ glass	230	360	3,390	1,910	1.89	6.9	18	0.27
70 $\text{Li}_2\text{S} \cdot 30\text{P}_2\text{S}_5$ glass	25	360	3,330	1,820	1.67	5.5	14	0.29
70 $\text{Li}_2\text{S} \cdot 30\text{P}_2\text{S}_5$ glass	240	360	4,020	2,090	1.91	8.3	22	0.31
75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	25	180	2,800	1,540	1.45	3.4	8.8	0.28
75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	25	360	3,520	1,870	1.68	5.9	15	0.30
75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	100	360	3,880	2,000	1.86	7.4	20	0.32
75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	150	360	4,040	2,060	1.89	8.0	21	0.32
75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	190	360	4,150	2,150	1.88	8.7	23	0.32
75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass-ceramic	230	360	4,110	2,090	1.88	8.2	22	0.33
80 $\text{Li}_2\text{S} \cdot 20\text{P}_2\text{S}_5$ glass	25	360	3,710	2,010	1.66	6.7	17	0.29
80 $\text{Li}_2\text{S} \cdot 20\text{P}_2\text{S}_5$ glass	25	540	3,920	2,130	1.73	7.9	20	0.29
80 $\text{Li}_2\text{S} \cdot 20\text{P}_2\text{S}_5$ glass	190	360	4,300	2,270	1.85	9.5	25	0.31
75 $\text{Na}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	25	360	3,280	1,690	1.87	5.3	14	0.32
75 $\text{Na}_2\text{S} \cdot 25\text{P}_2\text{S}_5$ glass	170	360	3,670	1,820	2.00	6.6	18	0.34
$\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$	—	—	5,510	2,760	4.52	34.0	92	0.33
$60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$	—	—	3,340	1,510	2.22	5.0	14	0.37

ature was investigated.

## 2. Experimental

The  $\text{Li}_2\text{S}$ – $\text{P}_2\text{S}_5$  glass solid electrolytes were prepared by mechanochemical milling in a planetary ball mill (Fritsch, Pulverisette 7). The starting materials of  $\text{Li}_2\text{S}$  (Idemitsu Kosan) and  $\text{P}_2\text{S}_5$  (Aldrich) were hand ground and the mixture was then placed into a zirconia ( $\text{ZrO}_2$ ) pots (internal volume of 45 ml) containing  $\text{ZrO}_2$  balls of 4 mm in diameter (500 balls).  $\text{Li}_2\text{S}$ – $\text{P}_2\text{S}_5$  glass powders were formed by a mechanical milling technique for 10 h at a rotation speed of the base disk of 510 rpm.  $\text{Na}_2\text{S}$ – $\text{P}_2\text{S}_5$  glasses were prepared from  $\text{Na}_2\text{S}$  (Nagao) by a similar procedure as that used to prepare the  $\text{Li}_2\text{S}$ – $\text{P}_2\text{S}_5$  glasses.<sup>5)</sup> All these processes were performed in a dry Ar atmosphere.

Prepared glass powders were compressed by conventional uniaxial cold press molding at room temperature or hot press molding at about 200°C to prepare 10-mm-diameter pellets. The molding time for cold and hot pressing was 5 min and 1 h, respectively. The densities of the glass sulfide electrolytes were determined by measuring the weight and volume of the prepared pellets. The volume of the samples for the calculation of density was directly measured from the geometry of prepared pellets in this study. The glass rod sample for compression test was prepared by uniaxial hot press with 360 MPa at 190°C for 1 h; the size was  $4 \times 4 \times 8$  mm. Compression stress–strain curves were obtained using strain rate of  $5 \times 10^{-3} \text{ s}^{-1}$ .

The Young's moduli were calculated using the following equations:

$$G = \rho V_S^2$$

$$\nu = (V_L^2 - 2V_S^2)/[2(V_L^2 - V_S^2)]$$

$$E = 2G(1 + \nu)$$

where  $V_S$  is velocity of shear wave,  $V_L$  is velocity of longitudinal wave,  $G$  is shear modulus,  $\rho$  is density of the pellet,  $\nu$  is the Poisson's ratio, and  $E$  is Young's modulus.

The Young's modulus was measured in plastic bags to avoid exposure to air for the sulfide glasses. The influence of polymer

bag was checked using the  $\text{Si}_3\text{N}_4$ ; the Young's modulus of  $\text{Si}_3\text{N}_4$  in the plastic bag was the same value to that without the bag. The influence of the bag was thus ignored because the bag was thin enough.

## 3. Results and discussion

The Young's moduli of the sulfide glass electrolyte pellets with various compositions, molding temperature and molding pressure were investigated. The Young's moduli measured using an ultrasonic sound velocity technique were listed in **Table 1**. The density of 50 $\text{Li}_2\text{S} \cdot 50\text{P}_2\text{S}_5$  glass prepared in this study ( $1.89 \text{ g cm}^{-3}$ ) is in good agreement with the density reported for the corresponding glass which was prepared by the melt quenching method,<sup>17)</sup> indicating that almost fully dense pellets were obtained by the hot press near 200°C. The Young's moduli of the  $\text{Li}_2\text{S}$ – $\text{P}_2\text{S}_5$  glasses prepared hot and cold pressing at 360 MPa were also shown in **Fig. 1**. The Young's moduli of  $\text{Li}_2\text{S}$ – $\text{P}_2\text{S}_5$  hot-pressed pellets were 18–25 GPa and those of cold-pressed pellets were about 14–17 GPa. The Young's moduli of cold-pressed pellets were lower than those of hot-pressed pellets. The Young's modulus of  $\text{Li}_2\text{S}$ – $\text{P}_2\text{S}_5$  glasses increases with increasing the  $\text{Li}_2\text{S}$  content in both hot-pressed and cold-pressed pellets. The highly dense pellets tend to be prepared by pressing with high molding temperature and pressure. For example, the densities of 75 $\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  glasses prepared at the pressures of 180 and 360 MPa at room temperature were 1.45 and  $1.68 \text{ g cm}^{-3}$ , respectively. Those prepared at room temperature, 100, 150 and 190°C at the 360 MPa were respectively 1.68, 1.86, 1.89, and 1.88  $\text{g cm}^{-3}$ . Almost fully dense pellets were obtained by hot press at temperatures higher than 100°C at 360 MPa. In this study, the Young's moduli of typical oxide electrolyte  $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$ <sup>18)</sup> and ionic glass  $60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$  were also evaluated. The  $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$  ceramic was prepared by normal sintering at a high temperature of 1230°C without pressure. The theoretical density of  $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$  calculated from crystal structure was  $5.1 \text{ g cm}^{-3}$ .<sup>19)</sup> The density of the prepared  $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$  pellet was  $4.52 \text{ g cm}^{-3}$ ; the relative density was 88%. The  $\text{Li}_7\text{La}_3\text{Zr}_2\text{O}_{12}$

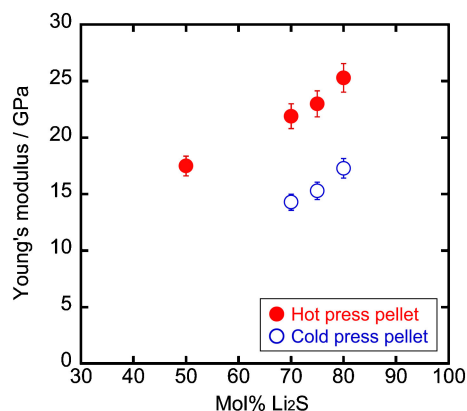


Fig. 1. Young's moduli of  $x\text{Li}_2\text{S} \cdot (100 - x)\text{P}_2\text{S}_5$  (mol %) glass pellets prepared by hot press and cold press at the pressure of 360 MPa.

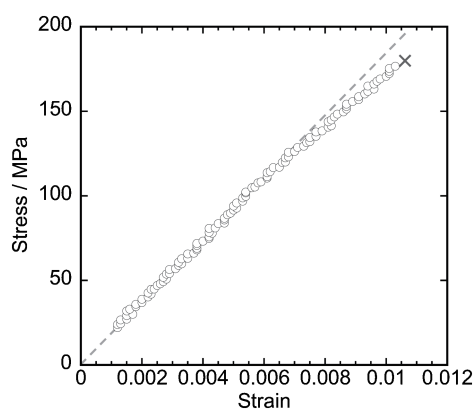


Fig. 2. Stress-strain curve in compression test for the  $75\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  hot press rod.

pellet showed a high Young's modulus of 92 GPa although it was not fully dense. The  $60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$  glass was prepared by the melt quenching method. The obtained glass was transparent and the density was  $2.22 \text{ g cm}^{-3}$ . The Young's modulus of  $60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$  was 14 GPa. The values were almost the same in a previous report.<sup>20)</sup>

The Young's modulus of a glass can be estimated from the bond dissociation energy per unit volume and the ion packing density.<sup>21)</sup> That is, glasses with large cations and/or anions tend to show low Young's modulus. Actually, the Young's modulus of  $75\text{Na}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  glass was smaller than that of  $75\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  glass; the size of  $\text{Na}^+$  ions is larger than  $\text{Li}^+$  ions. The coulomb energy between sulfur and sodium atoms in the  $75\text{Na}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  glass was smaller than that between sulfur and lithium atoms in the  $75\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  glass.

The  $60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$  glass showed similar Young's modulus to  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  solid electrolytes. The  $60\text{KNO}_3 \cdot 40\text{Ca}(\text{NO}_3)_2$  and  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  glasses are typical ionic glasses consisting large anions such as  $\text{NO}_3^-$ ,  $\text{PS}_4^{3-}$ , and  $\text{P}_2\text{S}_7^{4-}$ . As described above, existence of large anions and/or cations decreases Young's modulus of glasses.

The reasonability of the Young's modulus was also confirmed by the compression tests. Stress-strain curve in compression test for the  $75\text{Li}_2\text{S} \cdot 25\text{P}_2\text{S}_5$  glass rod is shown in Fig. 2. The glass rod was prepared by hot press; the size was  $4 \times 4 \times 8 \text{ mm}$ . The slope was straight below 130 MPa and slightly changes to be gentle above 130 MPa. The Young's modulus calculated based on the slope below 130 MPa of the stress-strain curve is ca. 17 GPa.

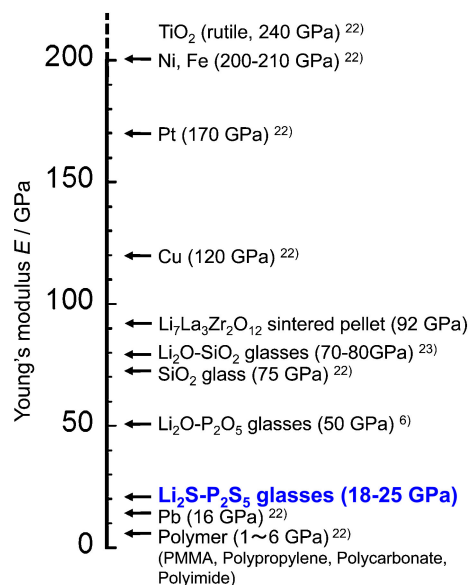


Fig. 3. Young's moduli of various materials.

The Young's modulus calculated from the stress-strain curve is similar to that measured by the ultrasonic sound velocity measurement (23 GPa). The change of the slope indicates that small degree of plastic deformation occurs before fracture. This phenomenon would relate to the high mechanical processability of  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  solid electrolytes. Further studies should be needed to clarify the details.

Young's moduli of various materials were shown in Fig. 3. The Young's moduli of  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  glasses are lower than those of oxide glasses and most transition metals and higher than polymers.<sup>6),22),23)</sup> Sulfides have lower bond dissociation energy per unit volume and lower ion packing density than oxides; sulfide glasses have thus lower Young's moduli than oxide glasses.

#### 4. Conclusion

The Young's moduli of  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  glassy solid electrolytes were measured by ultrasonic sound velocity measurements. The Young's moduli of the solid electrolytes were about 20 GPa. The Young's modulus increased with an increase of  $\text{Li}_2\text{S}$  content. The Young's moduli of the  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  glasses were lower than those of oxide based solid electrolytes. The Young's moduli of  $\text{Li}_2\text{S}-\text{P}_2\text{S}_5$  glasses prepared using various conditions were also measured. The pellets had high packing density even by pressing under glass transition temperature and showed Young's moduli close to the almost fully dense pellets. The reasonability of the value obtained by the ultrasonic measurement was confirmed by the compression test.

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