# Experimental Determination of Composite Stacking Sequence Using Four Point Bend Test

Nik Benko, John Callaway, Nick Dorsett, Martin Raming February 26, 2018

#### Abstract

There are several different techniques to determine ply orientations within a composite laminate. In this experiment we utilized laminated plate theory with experimental techniques in order to determine the correct stacking sequence of a composite laminate. We were provided a list of five possible stacking sequences for a composite panel in question. This panel was made of T800-3900 carbon fiber prepreg. Laminated plate thoery can be used to predict either stresses or strains of a composite laminate if given material properties, forces, and a stacking sequence. In this project we utilized laminated plate theory to predict strains for loading conditions that would match mechanical testing. Our mechanical testing consisted of a four-point bend test to induce a known moment in mid-section of our composite specimen. By attaching 3 strain gauges to the mid-section we recorded strains during mechanical testing. We then compared and best matched recorded strains to calculated strains to determine the correct stacking sequence.

### 1 Introduction

Composite materials are currently used in many applications where a high stiffness to weight ratio is important for design and operations such as in the aerospace and sports industries. While the idea of using combined materials to improve the capability of a structure has been utilized for thousands of years, use of modern composites like carbon fiber with an epoxy matrix is a relatively new practice. As the industry first began many questions developed involving engineering design implications such as failure criteria and material response. The main challenge for modern composites is the orthotropic nature of the material. Modern composite lamina tend to have a significantly larger Young's modulus in the fiber direction compared to the matrix direction. This orthotropic property is also what makes composites desirable when designing a structure that requires different material responses relative to global orientation. This is done by stacking individual plies in different orientations with respect to each other to create a laminate. The way in which these plies are oriented is also known as a stacking sequence.

To analytically understand how a stacking sequence of a laminate responds to a loading condition we turn to laminated plate theory (LPT). LPT first evolved in the 1950's and 60's.[?] The theory relies on the material properties in the material coordinates system, stacking sequence, and ply thickness to arrive at what is know as the ABD matrix. The ABD matrix can then be used to solve for strains, curvatures and stresses for a given load vector containing forces and moments. The ABD matrix is a  $6 \times 6$  matrix comprised of 36 components which makes for difficult hand calculations. For this reason it is often convenient to utilize a coding software to perform calculations as was done in this experiment.

LPT has proven to be very useful in design considerations but can be used for other purposes. In this report we explore the use and accuracy of LPT to back out a stacking sequence by comparing calculated results to experimental results. The goal of this experiment is to find which of the five unique laminate stacking sequence given matches the configuration of the specimen in question. In order to get data for comparison we loaded our specimen at intervals in a four-point bend flexure device while recording strains using strain gauges. We potentially only needed one induced moment but performed several for completeness and accuracy.

In this report we start by further expanding on LPT and implications of using the theory. We then discuss the experimental techniques and performed procedures used to arrive at force induced strains of our specimen. Possible errors and uncertainties relating to the experiment are then considered at the end of the methods section. Results are presented and discussed in the preceding our conclusion. All figures and tables can be located in section seven.

### 2 Methods

## 2.1 Laminated Plate Theory

The composite materials used in this lab are classified as specially orthotropic materials, meaning that they have the same material constants in two directions and differ in the third. This also means that four independent material constants are required to define the material stiffness. The constants used are E1, the stiffness in the fiber direction, E2, the stiffness 90 degrees from the fiber direction, G12, the shear modulus in the 1-2 plane, and v12, Poisson's ratio defining strain in the 2 direction due to applied force in the 1 direction. These material constants are used in Equation 1 to define the material stiffness matrix Q. This matrix in the 3x3 form shown assumes plane stress conditions, it was reduced from its 6x6 form by ignoring all z-direction stresses. This Q matrix is only applicable for a single ply with its fibers aligned in the x-direction. Tensor transformations can be used to apply stiffnesses in any orientation desired. Equation 2 shows the transformation matrix of sine and cosine terms used to transform the matrix, and equation 4 is the equation used to actually apply the rotation, generating  $Q_{bar}$ , used to denote this transformed stiffness. It is important to note that the Reuter matrix, denoted R, is used because its factor of 2 is necessary to satisfy tensor rotation rules.

These stiffnesses are used to calculate material constants for single plies of material. Laminated

Plate Theory (LPT) is used to compute matrices that relate forces, moments, strains, and curvatures. Using LPT requires the assumptions of perfectly bonded plies, plane sections remaining plane during deformation, thin laminates (length and width must be at least 10x the thickness), small mid-plane displacements and rotations, plane stress conditions, and transverse stresses to be neglected. These conditions being satisfied, equations 5, 6, and 7 are used to calculate three matrices denoted A, B, and D based up on the  $Q_{bar}$  matrix for each individual ply and z, which is the distance of each ply interface from the mid-plane of the laminate. These are combined into a 6x6 matrix that relates mid-plane strains and curvatures with normal forces and moments. The A matrix relates normal forces with normal strains, the D matrix relates moments and curvatures, and the B matrix relates both normal forces with curvatures and moments with normal strains.

#### 2.2 Experimental Techniques

#### Strain Analysis

Electrical strain gauges are a commonly used tool in engineering applications to accurately measure axial strain. The essential aspect of a strain gauge is made up of a very thin wire. Since the gauge is bonded to the surface of the material it extends or contracts with the specimen surface. This elongation or compression of the wires in the strain gauge causes the electrical resistance to change. Strain can then be determined by the fact that material deformation is dependent on material electrical resistivity. This relationship between resistivity and deformation of the given alloy in a strain gauge is referred to as the gauge facto, GF.

Electrical strain gauges are essentially a Wheatstone bridge which consists of four electrical resistors that act as two voltage dividers in parallel. There are three unique configurations that strain gauges can have depending on the number of active resistors in the Wheatstone bridge. These configurations are named full-bridge, half-bridge, and quarter-bridge of which contain four active elements, two active elements and one active element, respectfully. Additionally, the orientations of the active elements will determine the "type" of configuration. Testing setup, temperature differences, and type of strain to be measured dictates which configuration to use. In our experiment a quarter-bridge type I configuration was used for all strain gauges. This configuration was chosen since bending strain is to be measured and temperature effects will be negligible.

By applying a excitation voltage,  $V_{EX}$  to the strain gauges we can measure then voltage beyond the active resistors,  $V_{CH}$ .  $V_{CH}$  is typically very small so amplifiers are used in signal conditioning to boost signal levels. This aids in accuracy and resolution while reducing noise in the signal. Our amplification, Gain is calculated in equation 12 and is related to the  $V_{EX}$  and a scaling factor,  $R_{shunt}$  a setting in the Dewetron oscilloscope used for voltage measurement. From there we can calculate our voltage ratio,  $V_R$  with equation 11 and finally plug into equation 10 to convert voltage to strain.

Three strain gauges were implemented to measure strains near the mid section of the specimen. As figure ?? shows the strain gauges were oriented to achieve a 45° rosette. This type of strain rosette will allow for a direct measurement of strain in the lengthwise,  $\epsilon_x$  and strain in the crosswise direction,  $\epsilon_y$ . The shearing strain,  $\gamma_{xy}$  can then be calculated using equation 9

2.3 Procedure 2 METHODS

where,  $\epsilon_{OB}$  is placed at a 45° from either the crosswise or lengthwise axes.

#### **Mechanical Testing**

A four-point flexure test was chosen as the best mechanical testing procedure. As Figure?? shows the resultant moment is constant in the specimen within the region between the two loading points. Since the strain gauges are also located in this region we can be certain the measured strain is caused by the moment induced by the load frame. For each load the moment int his region was calculated with equation 13, where P is the load, and L is the span. Because there is inevitably some initial compliance within the load frame we chose the first load increment to be above 45 N to avoid false readings.

#### 2.3 Procedure

To begin the lab section, groups were given a 3"x12" strip of a composite laminate to test. To prepare it for testing, strain gauges were attached. To accomplish this, the surface of a glass plate, the sample, and a pair of tweezers were degreased with acetone. Next, the surface of the composite sample was wet sanded using 320 grit sandpaper and M-Prep Conditioner A. After sanding the surface was wiped clean. A square was used to make alignment marks on the surface of the sample with a pencil, making marks at 0, 45, and 90 degrees. M-Prep Neutralizer 5A was applied and the surface was wiped in a single direction with a fiber-free paper. This was continued until the paper did not have any contaminants on it after wiping. Once the sample surface was clean, the strain gauge was removed from its plastic sleeve and placed on the cleaned glass surface, bonding side down. Tape was placed on top of the strain gauge and used to lift it off the surface at a 45 degree angle or less. The strain gauge was then transferred onto the composite sample with the alignment marks on the gauge matched up with the lines drawn earlier. Once lined up, the tape was partly peeled up allowing access underneath the gauge region. M-Bond 200 catalyst was applied to the gauge region by lifting the brush out of the bottle, wiping the brush ten times against the inside of the neck of the bottle, and then sliding the brush over the gauge region, wiping all the way outside of the gauge region before lifting the brush. The catalyst was then allowed to dry for one minute. Within 3-5 seconds, one or two drops of M-Bond 200 adhesive was applied to the base of the tape, outside the gauge region and the tape, with the strain gauge, was pressed down onto the surface with a wiping motion, starting with the base of the tape where the adhesive was placed and moving along the length of the tape. Gentle pressure was then applied onto the top of the strain gauge for at lease one minute. Finally, the tape was removed by slowly peeling it directly back over itself. These steps were repeated twice more to attach all three strain gauges.

Following the attachment of the strain gauges, the lead wires from a Dewetron DEWE-30-8 were soldered to the terminals on the strain gauges by first priming the contacts with solder and then using a bead of solder to secure the wires to the terminals. The positive and negative terminal orientation did not matter. Once attached, the Wheatstone bridge corresponding to each gauge inside the DEWE-30-8 was balanced using the corresponding software, and the sample was gently flexed to ensure that the gauges were creating voltage differentials. Once everything was confirmed to be working, the sample was placed inside a four point

bend fixture with the guge region inside the load span on the fixture. Once everything was prepared, the sample was loaded at approximately 50 Newton increments. At each increment, the load was determined and the oscilloscope which was used to take voltage measurements was paused. The voltage outputs from the strain gauges were then recorded. This process was repeated using 50N increments until 250N and then the last load step was 30N to avoid over-stressing the laminate. Following this, the load was removed from the sample, and the sample was flipped over inside the fixture. The loading process was repeated with the laminate in this configuration. Finally, using the relevant equations, the voltages were related to strains and the forces applied were transformed into moments thereby allowing a list of possible laminates to be iterated across using a Matlab script until a match was found for the stacking sequence.

#### 2.4 Error and Uncertainties

For this experiment, there were two major sources of error and uncertainty. First was the alignment of the strain gauges. In order to get accurate in-plane strain measurements, the gauges in the rosette must be evenly spaced at 45° and aligned with the edges of the plate. Strain gauges placed by hand are inherently susceptible to mild misalignment. To quantify the deviation from 45° spacing, we imaged our plate after testing with a reference 90° angle and used ImageJ (NIH) to measure the angles between gauges. Figure (number) shows the image and corresponding alignment angles.

A second source of error came in the measurement of voltage. Voltage readings were taken by aligning the oscilloscope channel cursor with what was perceived to be the average of a voltage trace at a given time. As voltage increased, the sensitivity of the cursor decreased, limiting the ability to determine voltage to within more than 2-4% of the actual value. As a result, measurements of strain must also vary by  $\pm 4\%$ 

### 3 Results

Material properties of T800-3900 (Table 1), potential ply orientations (Table 2) and collected experimental data (Table 3) were combined using the previously described procedure to produce actual strains and predicted strains for the five potential layups. Results for normal strains in the x and y directions, along with shear strain are presented in Table 4.

Comparison of actual and predicted values show that Layup 3 ( $[(90)_2/(0)_4/(90)_2]_S$ ) was the composite laminate tested during the experimental process. Predicted vs. actual strains for this layup are presented in Figure 1.

## 4 Discussion

Determination of the ply orientation of the tested laminate was accomplished by eliminating potential stacking sequences from the provided list using a comparison of the actual measured strains with the predicted strains from laminated plate theory. Layups 2 and 5 were eliminated

by thickness of the laminate and large predicted strains in the x-direction compared to actual x-direction strains. Layup 1 was eliminated by large predicted shear strain values compared to actual shear strains. Layup 4 was eliminated by predicted positive shear strain compared to the actual negative shear strain seen during the experiment. This process resulted in the elimination of all layups with the exception of Layup 3

Although Layup 3 presented the best match of experimental and predicted strains, there are variations that can be seen, especially with the y-direction strains and the shear strains. As force was increased, the variation of actual strain and predicted strain increased. This can be explained by the errors in strain gauge orientation and voltage measurements described in the Error and Uncertainty section. This same error was not seen in the x direction, with actual values following the predicted values closely.

The variation seen in two of the three strain gauges shows one of the limitations of this method for determining stacking sequence. Since there is a potential for a high level of variation in strain measurements, this method is not sensitive to determining different stacking sequences that produce similar strain values. Also, this would provide potential prediction errors if a large number of stacking sequences are possible.

Another limitation of this method is that the possible stacking sequences must be known prior to performing the experiment. Carbon composite laminates provide an almost infinite number of potential stacking sequences. Even if the possible lamina orientations were limited to four (0,90,+45,-45), a 16 ply laminate (as was tested in this experiment) could have over 65,000 possible stacking sequences. It would be nearly impossible to eliminate all but one stacking sequence with the aforementioned error.

Despite the limitations of this method, it does effectively predict the stacking sequence of a carbon composite laminate when a small number of potential stacking sequences are known, and there is a large enough difference in the strain values predicted by laminated plate theory between these stacking sequences.

## 5 Conclusion

## 6 Figures

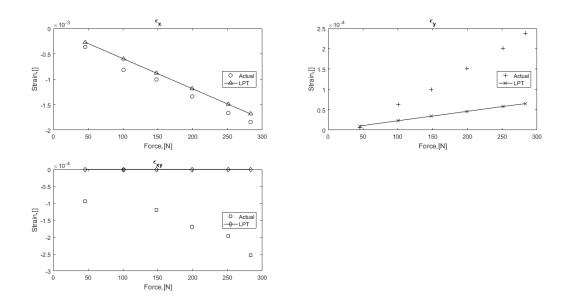


Figure 1: Force vs. Strain for Layup 3

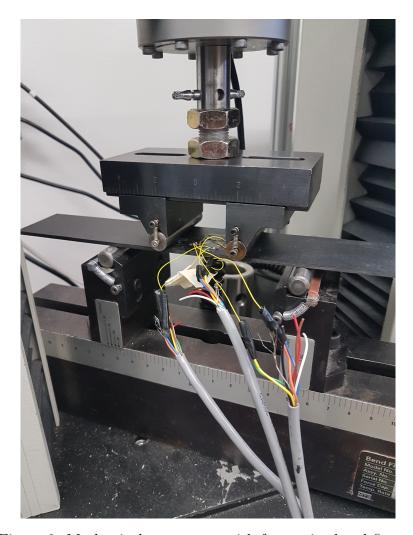


Figure 2: Mechanical test set up with four-point bend fixture  ${\cal P}$ 

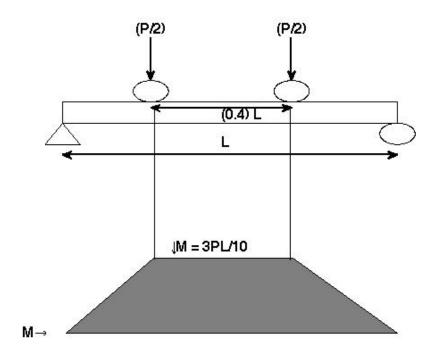


Figure 3: Moment diagram of four-point flexure setup used in mechanical testing.

## 7 Tables

Parameter	$\mathbf{E_1}$	$\mathbf{E_2}$	$G_{12}$	$\nu_{12}$	$\nu_{21}$
Value	114 GPa	8.3 GPa	3.93 GPa	0.33	0.02

Table 1: T800-3900 Material Properties

Layup #	Ply Orientation		
1	$[(90/45/0/45)_2]_S$		
2	$[45/90]_4$		
3	$[(90)_2/(0)_4/(90)_2]_S$		
4	$[90/45/0/-45]_4$		
5	$[(90)_2/(0)_2]_2$		

Table 2: Ply Orientation of Five Potential Layups

(degrees)

(degrees) (degrees)

(V)

(mv/V)

90

0

45

12

2.125 83.333

5.00002

Strain Gage Data

Gage 1 Orientation:

Support Spain.	0.121	(111)	dage i offentation.		
Load Span:	0.0508	(m)	Gage 2 Orientation:		
Length:	0.305	(m)	Gage 3 Orientation:		
Width:	0.04925	(m)			
Thickness:	0.0028	(m)	Excitation Voltage:		
			Gage Factor:		
			Dewetron Setting:		
			Calculated Gain:		
	Collected Loa	d and Voltage Da	ta		
	Gage	es on Top:			
Load (N)	Gage 1 Voltage	Gage 2 Voltage	Gage 3 Voltage		
45.78	-0.0002	0.0116	0.0072		
100.5	-0.002	0.026	0.012		
148.3	-0.0032	0.0322	0.0164		
199.5	-0.0048	0.0426	0.0216		
251	-0.0064	0.053	0.0264		
283	-0.0076	0.0588	0.0296		
		-			
()	Gages on Bottom:				
Load (N)	Gage 1 Voltage	Gage 2 Voltage	Gage 3 Voltage		
55.98	0.0044	-0.0092	-0.0064		
104.7	0.006	-0.0196	-0.0124		
160.7	0.0072	-0.0316	-0.0196		
204.8	0.0084	-0.0396	-0.0252		
256.5	0.008	-0.049	-0.032		
279.1	0.0104	-0.053	-0.0352		

0.127

(m)

Specimen Data

Support Span:

Table 3: Measured Data From Experimental Process

			$\epsilon_x$			
			$\mathbf{L}$	PT Predicte	ed	
Load (N)	Actual Strain	Layup 1	Layup 2	Layup 3	Layup 4	Layup 5
45.78	-0.000364	-0.000447	-0.004975	-0.000272	-0.000345	-0.001291
100.5	-0.000815	-0.000981	-0.010920	-0.000597	-0.000758	-0.002833
148.3	-0.001009	-0.001447	-0.016110	-0.000882	-0.001118	-0.004180
199.5	-0.001335	-0.001947	-0.021680	-0.001186	-0.001504	-0.005624
251	-0.001660	-0.002450	-0.027280	-0.001492	-0.001893	-0.007075
283	-0.001841	-0.002762	-0.030750	-0.001682	-0.002134	-0.007977
		$\epsilon_y$				
			L	PT Predicte	ed	
Load (N)	Actual Strain	Layup 1	Layup 2	Layup 3	Layup 4	Layup 5
45.78	0.000006	0.000041	0.000418	0.000010	0.000092	-0.000048
100.5	0.000063	0.000089	0.000918	0.000023	0.000203	0.000104
148.3	0.000100	0.000131	0.001354	0.000034	0.000299	0.000154
199.5	0.000151	0.000177	0.001822	0.000046	0.000403	0.000207
251	0.000201	0.000222	0.002292	0.000058	0.000507	0.000261
283	0.000238	0.000251	0.002584	0.000065	0.000571	0.000294
			$\epsilon_{xy}$		_	
			LPT Predicted			
Load (N)	Actual Strain	Layup 1	Layup 2	Layup 3	Layup 4	Layup 5
45.78	-0.000094	0.000308	0.003713	0.000000	0.000024	0.000000
100.5	0.000000	0.000675	0.008150	0.000000	0.000054	0.000000
148.3	-0.000120	0.000996	0.012030	0.000000	0.000079	0.000000
199.5	-0.000170	0.001340	0.016180	0.000000	0.000107	0.000000
251	-0.000196	0.001686	0.020360	0.000000	0.000134	0.000000
283	-0.000253	0.001901	0.022950	0.000000	0.000151	0.000000

Table 4: Predicted vs. Actual Strain Values for Potential Layups

## 8 Appendix

#### 8.1 Code

```
4 % LPT Code
5 %Text inputs to inputs.txt
6 %Line 1 is material properties E1, E2, G12, v12
7 %Line 2 is strength for failure critereon
8 %Line 3 is thermal coefficients
9 %Line 4 is ply thickness
10 %Line 5 is ply orientation layup
^{11} %Line 6 is loading - N<sub>-</sub>x, N<sub>-</sub>y, N<sub>-</sub>xy, M<sub>-</sub>x, M<sub>-</sub>y, M<sub>-</sub>xy
12 %Line 7 is temperature change
13
14 clear, clc
15 format short g
_{17} F = csvread ('inputs.txt');
18 fileID = fopen('inputs.txt');
%f %f %f %f %f %f %f', 'delimiter', ', ');
20 fclose (fileID);
for k = 1: length(C)
23 ply (k) = C\{k\}(5);
24 end
ply(find(isnan(ply))) = [];
27 % Name constants from text file:
28 % Material properties:
29 E1 = F(1,1);
30 E2 = F(1,2);
^{31} G = F(1,3);
v = F(1,4);
33
34 % Strength Properties:
S_L_plus = F(2,1);
_{36} S_L_{minus} = F(2,2);
_{37} S_{-}T_{-}plus = F(2,3);
_{38} S_{-}T_{-}minus = F(2,4);
39 S_{LT} = F(2,5);
41 % Coefficients of thermal expansion
alpha_1 = F(3,1);
alpha_2 = F(3,2);
45 % Ply thickness
t = F(4,1);
48 % Laminate stacking
```

```
_{49} L = ply;
50
51 % Forces and moments
_{52} N_{-x} = F(6,1);
N_y = F(6,2);
_{54} N_{xy} = F(6,3);
_{55} M_x = F(6,4);
_{56} M_{y} = F(6,5);
_{57} \text{ M}_{xy} = F(6,6);
59 % Temperature change
60 Temp = F(7,1);
61
62 %1. Calculate Q and S matrices:
63
_{64} R = \begin{bmatrix} 1 & 0 & 0 \end{bmatrix}
65 0 1 0
66 0 0 2];
68 S = [1/E1 - v/E1 0]
_{69} - v/E1 1/E2 0
70\ 0\ 0\ 1/G];
71
_{72} Q = inv(S);
74 % 1. Calculate Q_bar and S_bar matrices
for i = 1: length(L)
_{76} theta = L(i);
78 T = [(\cos d(theta)).^2 (\sin d(theta)).^2 2*\sin d(theta).*\cos d(theta)]
79 (\operatorname{sind}(\operatorname{theta})).^2 (\operatorname{cosd}(\operatorname{theta})).^2 -2*\operatorname{sind}(\operatorname{theta}).*\operatorname{cosd}(\operatorname{theta})
so-sind(theta).*cosd(theta) sind(theta).*cosd(theta) (cosd(theta)).^2-(sind(
        theta)).^2];
81
82 Q_{bar}(:,(i*3-2):i*3) = TQ*R*T/(R);
83 S_{bar}(:,(i*3-2):i*3) = inv(Q_{bar}(:,(i*3-2):i*3));
84
85 end
87 % Calculate Z values for plies:
so for i = 1: length(L) + 1
90 z(i) = -length(L)/2*t+((i-1)*t);
91 end
92
93 % 2. Calculate ABD matrix and inverse
95 A = zeros(3);
96 B = zeros(3);
property of D = zeros(3);
_{98} ABD = zeros(6);
100 for i= 1:3
for j = 1 : length(L)
```

```
A(i,1) = A(i,1) + Q_{-bar}(i,(j*3)-2)*(z(j+1)-z(j));
103 A(i,2) = A(i,2) + Q_{-bar}(i,(j*3)-1)*(z(j+1)-z(j));
A(i,3) = A(i,3) + Q_{-bar}(i,(j*3)-0)*(z(j+1)-z(j));
B(i,1) = B(i,1) + 0.5*(Q_{bar}(i,(j*3)-2)*(z(j+1)^2-z(j)^2));
B(i,2) = B(i,2) + 0.5*(Q_bar(i,(j*3)-1)*(z(j+1)^2-z(j)^2));
B(i,3) = B(i,3) + 0.5*(Q_{bar}(i,(j*3)-0)*(z(j+1)^2-z(j)^2));
D(i,1) = D(i,1) + (1/3) * (Q_{bar}(i,(j*3)-2) * (z(j+1)^3-z(j)^3));
109 D(i,2) = D(i,2) + (1/3) * (Q_{bar}(i,(j*3)-1) * (z(j+1)^3-z(j)^3));
110 D(i,3) = D(i,3) + (1/3) * (Q_{bar}(i,(j*3)-0) * (z(j+1)^3-z(j)^3));
111 end
112
   end
114 % make values 0 that should be 0:
115 for i = 1:3
116 for j = 1:3
if A(i,j) < 1*10^-12 \&\& A(i,j) > -1*10^-12
118 A(i,j) = 0;
119 end
if B(i,j) < 1*10^-12 \&\& B(i,j) > -1*10^-12
121 B(i,j) = 0;
122 end
if D(i,j) < 1*10^-12 \&\& D(i,j) > -1*10^-12
124 D(i,j) = 0;
125 end
126 end
   end
127
ABD(1:3,1:3) = A;
ABD(4:6,1:3) = B;
ABD(1:3,4:6) = B;
ABD(4:6,4:6) = D;
ABDinv = ABD^-1;
135 % 3. Apparent laminate stiffness properties:
_{136} E_{-x} = 1/(length(L)*t*ABDinv(1,1));
E_y = 1/(length(L)*t*ABDinv(2,2));
   v_x = -ABDinv(1,2)/ABDinv(1,1);
   G_xy = 1/(length(L)*t*ABDinv(3,3));
   E_fx = \frac{12}{((length(L)*t)^3*ABDinv(4,4))};
   E_{fy} = 12/((length(L)*t)^3*ABDinv(5,5));
141
142
143 % 4. Midplane strains and curvatures
144 % Calculate thermal expansion coefficients in x-y:
145 % Calculate N and M thermal
alpha = [alpha_1, alpha_2, 0];
_{147} \text{ N}_{-}\text{T} = [0, 0, 0]';
_{148} \text{ M}_{-}\text{T} = [0, 0, 0]';
for i = 1: length(L)
   theta = L(i);
150
_{152} T = [(\cos d(theta)).^2 (\sin d(theta)).^2 2*\sin d(theta).*\cos d(theta)]
153 (sind(theta)).^2 (cosd(theta)).^2 -2*sind(theta).*cosd(theta)
-\sin d (theta) \cdot *\cos d (theta) \sin d (theta) \cdot *\cos d (theta) (\cos d (theta)) \cdot 2 - (\sin d (theta)) \cdot 2 - (theta) \cos d (theta)
       theta)).^2];
```

```
alpha_xy = inv(T)*alpha;
Q_bar_T = TQ*R*T/(R);
N_T = N_T + Temp * Q_bar_T * alpha_xy * (z(i+1)-z(i));
M_T = M_T + 0.5 * Temp * Q_bar_T * alpha_xy * (z(i+1)^2 - z(i)^2);
160 N = [N_x, N_y, N_{xy}]' + N_T;
161 M = [M_x, M_y, M_{xy}]' + M_T;
_{162} \text{ NM} = [N]
163 M];
         strain_mid = ABDinv*NM;
165
eps_x_mid = strain_mid(1);
         eps_y_mid = strain_mid(2);
eps_xy_mid = strain_mid(3);
_{170} \text{ K_x} = \text{strain_mid}(4);
_{171} \text{ K_y} = \text{strain_mid}(5);
        K_xy = strain_mid(6);
173
174 % 5. Calculation of strains at all points in laminate and plot
176 % Calculation of mechanical strain
        eps_mech_x = eps_x_mid + z*K_x;
         eps_mech_y = eps_y_mid + z*K_y;
         eps_mech_xy = eps_xy_mid + z*K_xy;
180
         eps_mech = [eps_mech_x;eps_mech_y;eps_mech_xy];
181
182
        figure (1)
         plot (eps_mech_x, z, 'r-', eps_mech_y, z, 'b-', eps_mech_xy, z, 'g-')
        set(gca, 'ydir', 'reverse')
legend('\epsilon_x','\epsilon_y','\epsilon_x_y')
         grid on
        xlabel ('Strain, []')
188
         ylabel('z,[m]')
          title ('Part 5 - Strain')
192 % 6. Calculation of stresses at top and bottom of ply:
194 % Get repeating z values at ply interfaces
195 for i = 1 : length(L) + 1
        zplot((i*2)-1)=z(i);
zplot(i*2)=z(i);
        end
198
        zplot(2*(length(L)+1)) = [];
199
         zplot(1) = [];
201
for i = 1: length(L)
theta = L(i);
T = [(\cos d(theta)).^2 (\sin d(theta)).^2 + \sin d(theta).*\cos d(theta)]
206 (sind(theta)).^2 (cosd(theta)).^2 -2*sind(theta).*cosd(theta)
207 - \text{sind}(\text{theta}) \cdot * \cos d(\text{theta}) \sin d(\text{theta}) \cdot * \cos d(\text{theta}) \cdot (\cos d(\text{theta})) \cdot 2 - (\sin d(\text{the
                    theta)).^2];
```

```
alpha_xy = inv(T)*alpha;
eps_therm = alpha_xy*Temp;
eps1 = eps\_mech(:, i)-eps\_therm;
eps2 = eps\_mech(:, i+1)-eps\_therm;
stress\_top(:,i) = Q\_bar(:,3*i-2:3*i)*(eps1);
   stress\_bottom(:,i) = Q\_bar(:,3*i-2:3*i)*(eps2);
214
   end
215
for i = 1: length(L)
stress_xy(:, i*2-1) = stress_top(:, i);
stress_xy(:,i*2) = stress_bottom(:,i);
219
  end
220
221 % 7. Plots of global stresses for each ply:
222
223 figure (2)
224 plot (stress_xy (1,:), zplot, '-*')
225 legend('\sigma_x')
set (gca, 'ydir', 'reverse')
227 grid on
228 xlabel('\sigma_x, [Pa]')
  ylabel('z,[m]')
title('\setminus sigma_x')
231
232 figure (3)
233 plot (stress_xy (2,:), zplot, '-*')
234 legend('\sigma_y')
set (gca, 'ydir', 'reverse')
236 grid on
237 xlabel('\sigma_y, [Pa]')
238 ylabel ('z, [m]')
title('\sigma_y')
figure (4)
242 plot (stress_xy (3,:), zplot, '-*')
243 legend('\sigma_x_y')
set (gca, 'ydir', 'reverse')
245 grid on
xlabel(' \setminus sigma_x, [Pa]')
  ylabel('z, [m]')
   title ('\sigma_x_y')
248
249
250 % 8. Material coordinates on top and bottom of each ply
for i = 1: length(L)
  theta = L(i);
254
^{255} T = [(\cos d(theta)).^2 (\sin d(theta)).^2 2*\sin d(theta).*\cos d(theta)]
   (\sin d(\text{theta})).^2 (\cos d(\text{theta})).^2 -2*\sin d(\text{theta}).*\cos d(\text{theta})
  -\sin d (theta).*\cos d (theta) \sin d (theta).*\cos d (theta) (\cos d (theta)).^2-(\sin d (
       theta)).^2];
alpha_xy = inv(T)*alpha;
eps_therm = alpha_xy*Temp;
```

```
eps1 = eps\_mech(:, i)-eps\_therm;
eps2 = eps\_mech(:, i+1)-eps\_therm;
  stress\_top(:,i) = Q\_bar(:,3*i-2:3*i)*(eps1);
stress\_bottom(:,i) = Q\_bar(:,3*i-2:3*i)*(eps2);
  stress\_top\_12(:,i) = T*stress\_top(:,i);
   stress\_bottom\_12(:,i) = T*stress\_bottom(:,i);
  end
267
268
for i = 1: length(L)
  stress_12(:, i*2-1) = stress_top_12(:, i);
  stress_12(:,i*2) = stress_bottom_12(:,i);
  end
273
274 % 9. Hashin Failure Criteria:
275
276 Fiber_Failure = ((stress_12(1,:).^2)./(S_L_plus*S_L_minus))+...
   ((1/S_L_plus - 1/S_L_minus).*stress_12(1,:))+...
   ((stress_12(3,:).^2)./(S_LT^2));
  Resin_Failure = ((stress_12(2,:).^2)./(S_T_plus*S_T_minus))+...
   ((1/S_T_plus - 1/S_T_minus).*stress_12(2,:)) + ...
   ((stress_12(3,:).^2)./(S_LT^2));
283 mode = 'No Failure';
^{284} M=0;
285
286 i = 1;
while i \leq length(L)
288 theta2 ((i*2)-1)=L(i);
289 theta2 (i*2)=L(i);
290 i = i+1;
  end
291
292
  if max(Fiber_Failure) >= 1
mode = 'Fiber Failure';
   [M, I] = \max(Fiber_Failure);
  z_f ail_f = zplot(I);
   ply_fail_f = theta2(I);
297
  end
  if max(Resin_Failure) >= 1
   [M2, I2] = \max(Resin_Failure);
   if M2 > M
301
302 mode = 'Resin Failure';
z_fail_r = zplot(I2);
   ply_fail_r = theta2(I2);
305 end
зов end
307
308 % Outputs to .txt file:
309 i = 1;
fid = fopen( 'outputs_JohnCallaway.txt', 'wt');
311 fprintf(fid , 'John Callaway \n');
s12 fprintf(fid , 'LPT Code - Spring 2017 \n');
sis fprintf(fid , '\n Part 1 \n');
314 for i = 1: length(L)
```

```
fprintf (fid, '\n Ply %i: \n',i);
s16 fprintf(fid , '\n The Q matrix is:\n');
  fprintf(fid, '%8.3g %8.3g \%8.3g\n',Q);
   fprintf(fid , '\n The Q bar Matrix is:\n');
   fprintf(fid, '%8.3g %8.3g %8.3g\n', Q_bar(:, i*3-2:i*3));
   fprintf(fid , '\n The S matrix is:\n');
   fprintf(fid, '%8.3g %8.3g \%8.3g\n',S);
   fprintf(fid , '\n The S bar Matrix is:\n');
   fprintf(fid, '%8.3g %8.3g %8.3g\n',S_bar(:,i*3-2:i*3));
  end
   fprintf(fid, '\n Part 2: \n');
325
   fprintf(fid, '\n The ABD matrix is: \n');
   fprintf(fid, '%8.4g %8.4g %8.4g %8.4g %8.4g \n',ABD);
   fprintf(fid , '\n The ABD inverse matrix is: \n');
   fprintf(fid, '%8.4g %8.4g %8.4g %8.4g %8.4g \n', ABDinv);
   fprintf(fid, '\n Part 3: \n');
   fprintf(fid , '\n E_x is: \n');
  fprintf(fid, '%8.2g \n', E_x);
   fprintf(fid , '\n E_y is: \n');
   fprintf(fid, '%8.2g \n', E_y);
   fprintf(fid , '\n v_xy is: \n',);
   fprintf(fid, '%8.2g \n', v_xy);
   fprintf(fid , '\n G_xy is: \n');
   fprintf(fid, '\%8.2g \ \ \ \ G_xy);
   fprintf(fid , '\n E_fx is: \n');
340
   fprintf(fid , '%8.2g \n', E_fx);
   fprintf(fid, '\n E_fy is: \n');
   fprintf(fid , '%8.2g \n', E_fy);
344
   fprintf(fid, '\n Part 4: \n');
  fprintf(fid , '\n Midplane Strain in x is: \n');
   fprintf(fid, \%8.4g \ n', eps_x_mid);
   fprintf(fid , '\n Midplane Strain in y is: \n');
   fprintf(fid , '%8.4g \n', eps_y_mid);
   fprintf(fid , '\n Midplane Strain in xy is: \n');
   fprintf(fid , '%8.4g \n', eps_xy_mid);
   fprintf(fid , '\n Curvature in x is: \n');
   fprintf(fid , '\n Curvature in y is: \n');
   fprintf(fid, '%8.4g \n', K_y);
   fprintf(fid , '\n Curvature in xy is: \n');
   fprintf(fid , '%8.4g \n', K_xy);
357
   fprintf(fid, '\n Part 6: \n');
359
   fprintf(fid , '\n Global Coordinate Stresses: \n');
  for i = 1: length(L)
   fprintf(fid , '\n Ply %i Top: ',i);
   fprintf(fid, '%8.4g %8.4g %8.4g \n', stress_top(:,i));
   fprintf(fid , '\n Ply %i Bottom: ',i);
   fprintf(fid, '%8.4g %8.4g \n', stress_bottom(:,i));
  end
366
367
368 fprintf(fid, '\n Part 8: \n');
```

8.2 equations 8 APPENDIX

```
369 fprintf(fid, '\n Material Coordinate Stresses: \n');
370 for i = 1: length(L)
fprintf(fid, '\n Ply %i Top:',i);
fprintf(fid, ' %8.4g %8.4g %8.4g \n', stress_top_12(:,i));
fprintf(fid, '\n Ply %i Bottom: ', i);
   fprintf(fid , ' %8.4g %8.4g %8.4g \n', stress_bottom_12(:,i));
375
   end
376
   fprintf(fid, '\n Part 9: \n');
377
379 if strcmp (mode, 'No Failure')
380 fprintf(fid, '\n No failure occured \n');
382 if strcmp (mode, 'Fiber Failure')
383 fprintf (fid, 'Ply %i failed first, its orientation is %d degrees n', ceil (I/2),
       ply_fail_f);
fprintf(fid, 'The location of the failure is %8.3g m \n', z_fail_f');
fprintf(fid, 'The failure occured in the fibers \n');
386 end
387 if strcmp (mode, 'Resin Failure')
388 fprintf(fid, 'Ply %i failed first, its orientation is %d degrees \n', ceil(I2
      /2), ply_fail_r);
sse fprintf(fid, 'The location of the failure is %8.3g m \n',z_fail_r');
390 fprintf(fid, 'The failure occured in the resin \n');
391 end
```

#### 8.2 equations

$$[Q] = \begin{bmatrix} \frac{E_1}{1 - \nu_{12}\nu^{21}} & \frac{\nu_{12}E_2}{1 - \nu_{12}\nu^{21}} & 0\\ \frac{\nu_{12}E_2}{1 - \nu_{12}\nu^{21}} & \frac{E_2}{1 - \nu_{12}\nu^{21}} & 0\\ 0 & 0 & \frac{1}{G_{12}} \end{bmatrix}$$
(1)

$$[T] = \begin{bmatrix} \cos^2(\theta) & \sin^2(\theta) & 2\sin(\theta)\cos(\theta) \\ \sin^2(\theta) & \cos^2(\theta) & -2\sin(\theta)\cos(\theta) \\ -\sin(\theta)\cos(\theta) & \sin(\theta)\cos(\theta) & \cos^2(\theta) - \sin^2(\theta) \end{bmatrix}$$
(2)

$$[R] = \begin{bmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 2 \end{bmatrix} \tag{3}$$

$$\bar{Q} = T^{-1}QRTR^{-1} \tag{4}$$

8.2 equations 8 APPENDIX

$$A_{ij} = \sum_{k=1}^{N} (\bar{Q}_{ij})_k (z_k - z_{k-1})$$
 (5)

$$B_{ij} = \sum_{k=1}^{N} (\bar{Q}_{ij})_k (z_k^2 - z_{k-1}^2)$$
(6)

$$D_{ij} = \sum_{k=1}^{N} (\bar{Q}_{ij})_k (z_k^3 - z_{k-1}^3)$$
 (7)

$$\begin{bmatrix} N_x \\ N_y \\ N_{xy} \\ M_x \\ M_y \\ M_{xy} \end{bmatrix} = \begin{bmatrix} A_{11} & A_{12} & A_{16} & B_{11} & B_{12} & B_{16} \\ A_{21} & A_{22} & A_{26} & B_{21} & B_{22} & B_{26} \\ A_{61} & A_{62} & A_{66} & B_{61} & B_{62} & B_{66} \\ B_{11} & B_{12} & B_{16} & D_{11} & D_{12} & D_{16} \\ B_{21} & B_{22} & B_{26} & D_{21} & D_{22} & D_{26} \\ B_{61} & B_{62} & B_{66} & D_{61} & D_{62} & D_{66} \end{bmatrix} \begin{bmatrix} \epsilon_x \\ \epsilon_y \\ \epsilon_{xy} \\ \kappa_x \\ \kappa_y \\ \kappa_{xy} \end{bmatrix}$$

$$(8)$$

$$\gamma_{xy} = 2\epsilon_{OB} - (\epsilon_x + \epsilon_y) \tag{9}$$

$$\epsilon = \frac{-4V_R}{GF(1+2V_R)}\tag{10}$$

$$V_R = \frac{V_{CH} - V_0}{V_{EX}Gain} \tag{11}$$

$$Gain = \frac{5}{\left(\frac{V_{EX}}{1000}\right)R_{shunt}} \tag{12}$$

$$M = \frac{3PL}{20} \tag{13}$$