

DEEPER INSIGHTS  
UNVEILING DOPANT PROFILES WITH GIXRD

by

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DEPARTMENT APPROVAL

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This thesis has been reviewed by the research advisor, research coordinator, and department chair and has been found to be satisfactory.

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## ABSTRACT

## DEEPER INSIGHTS

### UNVEILING DOPANT PROFILES WITH GIXRD

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In this work, grazing incidence X-ray diffraction (GIXRD) is utilized to investigate the depth profile of dopants in silicon. By systematically varying the X-ray incidence angle, one can precisely control the depth penetrated into doped silicon samples. This approach enables a non-destructive analysis of how dopant concentration evolves with depth. The findings provide crucial insights into the distribution and behavior of dopants, which is essential for optimizing semiconductor device performance and fabrication processes.

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# Chapter 1

## Background

Solids can be divided into two broad types: crystalline, and amorphous. Crystalline solids have repeating structure that is predictable, and is consistent throughout the entire solid. Amorphous solids have no repeating structures, and have no preferred orientation.

X-Ray diffraction (XRD) utilizes x-rays to non-destructively probe the inner atomic layers of materials. In this work it is used to probe different depths of doped silicon to see dopant levels as a function of depth. As the x-ray interacts with the sample it will reflect, refract, and diffract. Diffraction is the process of note, and the only one that will be investigated for this work.

When XRD is used on crystalline solids a diffraction pattern will emerge. The spacing between the atoms acts as the slit size that allows for the diffraction interference pattern, in accordance with Bragg's Law ( $2d \sin \theta = n\lambda$ ). There also exist different spacings between atoms that are not nearest neighbors. Each of these different spacings are relegated into planes that act like a diffraction grating, existing for every possible repeating spacing within the atom. All of these in superposition produce a graph that shows sharp peaks correlating directly to the atomic plane's

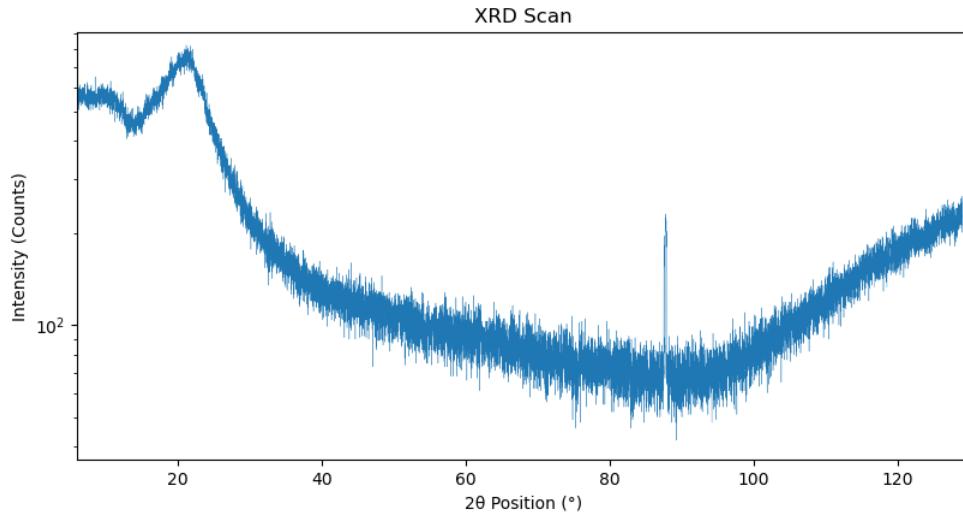
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spacing (shown in figure 1.1).



**Figure 1.1** Diffractogram of pure crystalline solid

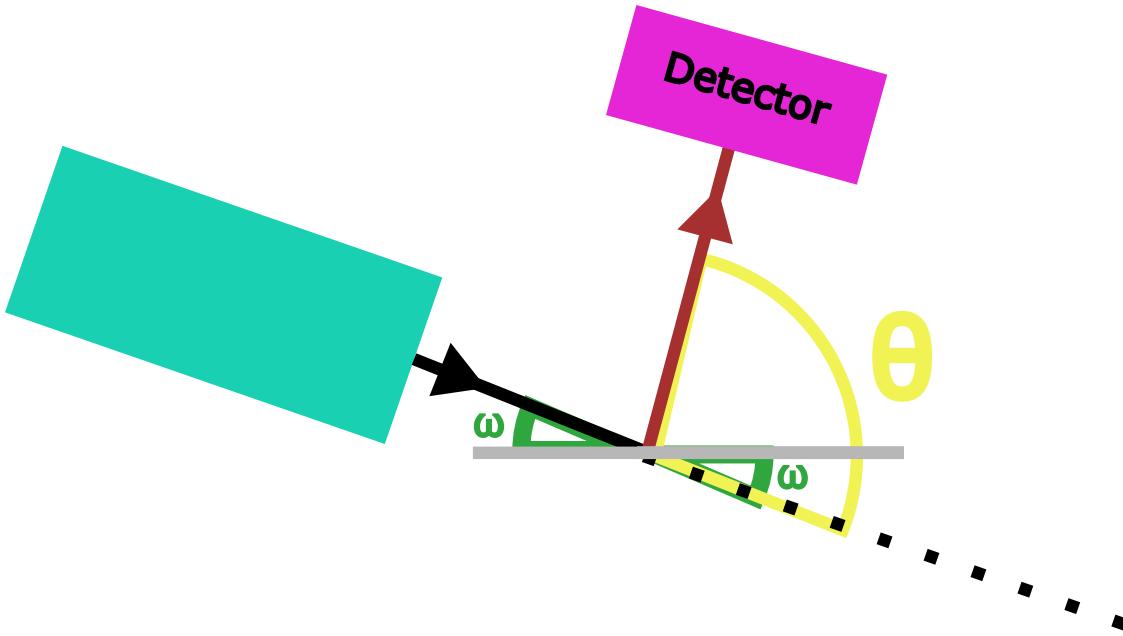
Amorphous solids under the x-ray beam behave in a similar manner. The difference being because every atom is arranged randomly, there are an infinite number of planes with their own spacing. As shown in figure 1.2, this correlates to a moderately low angle hump (called an amorphous hump) with all spacings that would correlate to higher angled peaks being nonexistent because of deconstructive interference[1–3].



**Figure 1.2** Diffractogram of pure amorphous solid

The materials analyzed in this work are silicon doped with phosphorous, silicon doped with boron, and pure silicon. The first two materials will be best thought of as a gradient from pure phosphorous or boron to pure silicon from top to bottom. Silicon is a crystalline material, while phosphorous and boron are amorphous. This means that the diffractogram will be a superposition of an amorphous diffractogram on a crystalline diffractogram. The silicon sample was only used to verify that the amorphous hump was present on the diffractograms.

The XRD process used to incrementally probe deeper depths is called grazing incidence x-ray diffraction (GIXRD, also called glancing incidence x-ray diffraction). Using this method the XRD will hold the incident beam at a fixed angle (called omega ( $\omega$ )), while the diffracted beam side moves from a starting angle to an end angle. The detector is used in a 0D mode that simply counts all the x-rays that interact with it. The angle the counts correspond to is the angle the incident beam is added to the current diffracted beam angle all multiplied by 2:  $2(\omega + \theta)$ . After the scan finishes a new scan is started with a different incident angle to probe a different depth.



**Figure 1.3** Diagram of XRD using GIXRD

The different incident angles probe different depths due to the different path lengths they provide the x-rays as they interact with the samples. This is due to the exponential attenuation of intensity as the beam travels given by this equation:  $I(y) = I_0 e^{-\alpha y}$  with  $\alpha = 2\omega n_I / c$  [4, 5]. With the complex index of refraction ( $n_I$ ) strongly dependent on wavelength. The x-ray source is copper, producing  $K_\alpha$ , and  $K_\beta$  in most abundance. Since one wavelength provides the most uniformity on the diffractogram, the incident beam optics include a filter to limit the unnecessary wavelength.  $K_\alpha$  is what is most common, so the  $K_\beta$  gets mostly filtered out by a nickel filter. This results in a x-ray beam that has a wavelength of  $1.54\text{\AA}$  (uniformity in the wavelength is also necessary for Bragg's law).

# Chapter 2

## Methods

Three different methods for determining relative percentages of the dopant in the substrate are used. The methods are named: 1. Intensity, 2. Curve Fitting, and 3. Summation. Before discussing the intricacies of these methods, it is important to first know what shared simplifying assumptions on which they are predicated. The first of these assumptions is that the mixture of the probed depth is uniform. Uniformity of the mixture cannot be achieved in the sample due to the diffusion process starting at the top of the sample working its way down. The next assumption is that the intensity of the x-ray has been attenuated by 90%. The thirdly, that the sample is put in its preferred orientation that allows for the silicon structure to be aligned with the x-ray beam. The last assumption is the index of refraction is the same as bulk samples[6] as those values will be utilized.

### **Method 1: Intensity**

This method takes the highest count of the amorphous hump, and the highest count of the crystalline peaks as input parameters, to take the ratio of them. The following

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formula is used for the calculation:

$$m = \frac{I_a}{I_a + I_c}$$

$I_a$  is the highest count of the amorphous hump,  $I_c$  is the highest count of the crystalline peaks, and  $m$  is the mixture percent.

### Method 2: Curve Fitting

The amorphous hump, and the crystalline peaks can be approximated by Gaussian curves. As such, it is possible to fit a curve to the hump, and the desired peak. With this fit, analytic integration is performed. The values from the integrations are then taken as a ratio as in the intensity method. This method also gives a background baseline that can be subtracted out.

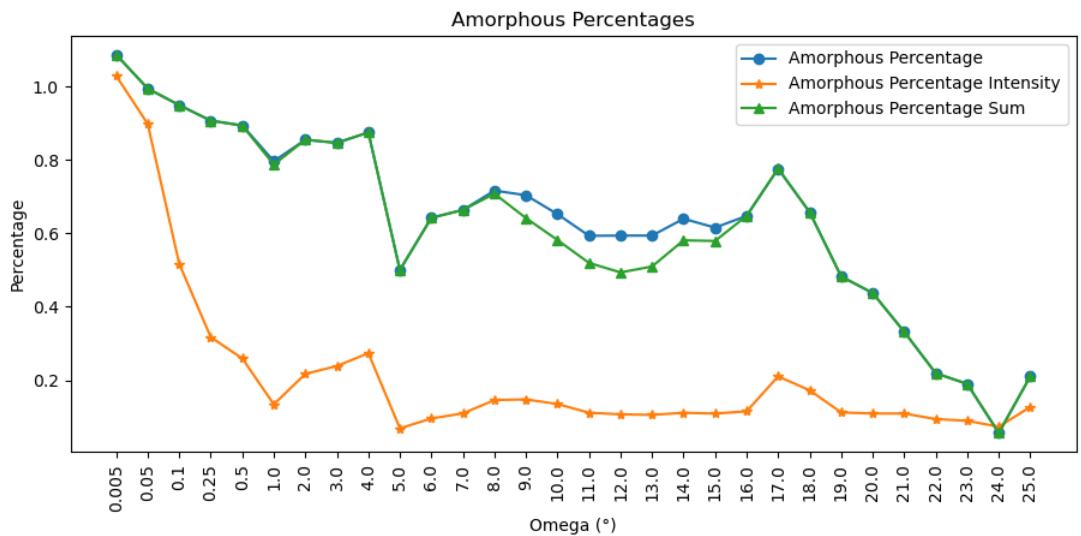
### Method 3: Summation

This method is similar to the curve fitting method, but instead of fitting the hump, and peak to a Gaussian, a numerical integration is performed. This numerical integration sums up the heights in the region of the relevant area to take their ratio. As stated, only the heights are summed; This is due to the spacing between all points being identical. Below is the proof:

$$\begin{aligned} m &= \frac{\sum_{\theta_i}^{\theta_f} y_a dx_a}{\sum_{\theta_i}^{\theta_f} y_a dx_a + \sum_{\theta_i}^{\theta_f} y_c dx_c} \\ dx_a &= dx_c = \text{Constant} \\ m &= \frac{dx_a \sum_{\theta_i}^{\theta_f} y_a}{dx_a \sum_{\theta_i}^{\theta_f} y_a + dx_c \sum_{\theta_i}^{\theta_f} y_c} \\ m &= \frac{dx_a \sum_{\theta_i}^{\theta_f} y_a}{dx_a \left( \sum_{\theta_i}^{\theta_f} y_a + \sum_{\theta_i}^{\theta_f} y_c \right)} \\ m &= \frac{\sum_{\theta_i}^{\theta_f} y_a}{\sum_{\theta_i}^{\theta_f} y_a + \sum_{\theta_i}^{\theta_f} y_c} \end{aligned}$$

# Chapter 3

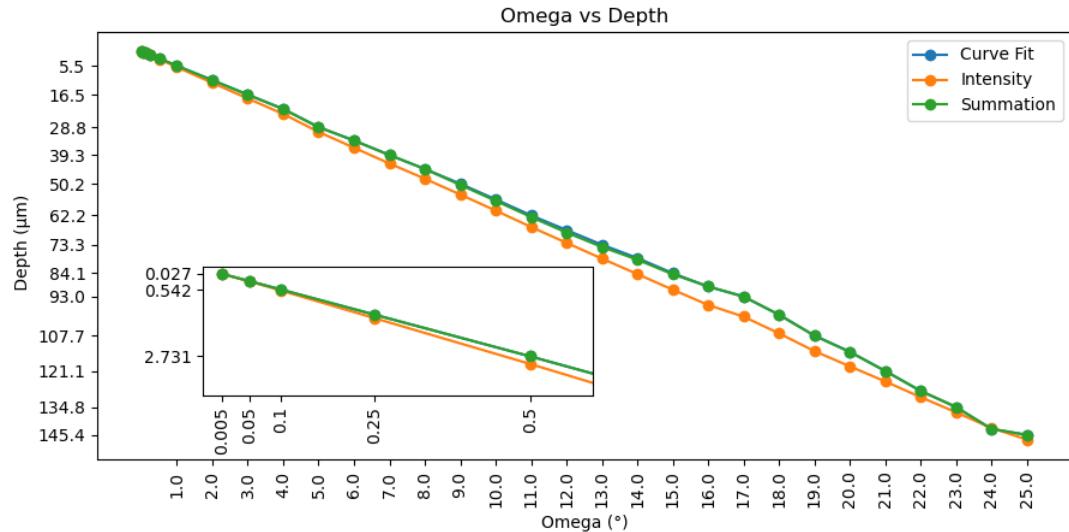
## Results



**Figure 3.1** Percent amorphous at selected omegas

Figure 3.1 provides a concise overview of how the dopant and substrate mixture change with omega. All three methods show the amorphous percent decreases with depth, which is to be expected. They also show that the 0.005 omega is greater than 1 for the amorphous. This is due to the crystalline peak being nonexistent, but the program is still trying to see a peak. The curve fitting and summation method provide excellent agreement with each other, while the ratio method alludes to an

exponential.



**Figure 3.2** Depth vs omega for the three methods

Figure 3.2 shows the depth that correlates with the three methods. The methods have a close agreement, with the curve fit and summation method being right on top of each other. They all show a mostly linear relationship, which is to be expected since the indices of refraction are very similar for the dopant and substrate. Nevertheless, it is interesting that the methods more closely agree at lower and high omegas, but not the middle omegas.

# **Chapter 4**

## **Analysis**

# **Chapter 5**

## **Conclusion**

Using GIXRD to determine relative composition of samples led to three analysis methods. Two of the methods utilized integration, and their results are closer aligned. One method used peak intensity ratios, and provides results that noticeably differ. All three methods provide reasonable agreement as to the depth that each omega probed. Using GIXRD to see relative mixture amount dependent on depth has been shown to be useful, keeping in mind the limitations given by the assumptions.

### **5.1 Future Work**

Performing this analysis on a known sample would be able to provide greater validity to the results of this analysis. Creating an analysis to correct for the real diffusion process by not assuming uniform layers. This would produce the greatest impact on the real world application of these results.

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# **Appendix A**

## **XRD Manual**

The purpose of an appendix is to provide supplementary information which would distract if included in the main body of the thesis. Items appearing as an appendix might include lengthy derivations. If students feel compelled to include a brief tutorial on relevant background information (not new research), it should appear as an appendix. An appendix might consist of portions of unique computer code that was developed as part of the project.