

RCW Report

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Abstract

In this work, grazing incidence X-ray diffraction (GIXRD) is utilized to investigate the depth profile of dopants in silicon. By systematically varying the X-ray incidence angle, one can precisely control the depth penetrated into doped silicon samples. This approach enables a non-destructive analysis of how dopant concentration evolves with depth. The findings provide crucial insights into the distribution and behavior of dopants, which is essential for optimizing semiconductor device performance and fabrication processes.

1 Background

Solids can be divided into two broad types: crystalline, and amorphous. Crystalline solids have repeating structure that is predictable, and is consistent throughout the entire solid. Amorphous solids have no repeating structures, and have no preferred orientation.

X-Ray diffraction (XRD) utilizes x-rays to non-destructively probe the inner atomic layers of materials. In this work it is used to probe different depths of doped silicon to see dopant levels as a function of depth. As the x-ray interacts with the sample it will reflect, refract, and diffract. Diffraction is the process of note, and the only one that will be investigated for this work.

When XRD is used on crystalline solids a diffraction pattern will emerge. The spacing between the atoms acts as the slit size that allows for the diffraction interference pattern, in accordance with Bragg's Law ($2d \sin \theta = n\lambda$). There also exist different spacings between atoms that are not nearest neighbors. Each of these different spacings are relegated into planes that act like a diffraction grating, existing for every possible repeating spacing within the atom. All of these in superposition produce a graph that shows sharp peaks correlating directly to the atomic plane's spacing (shown in figure 1).

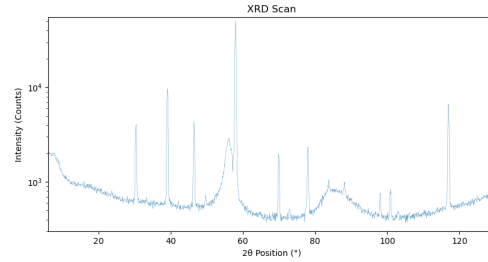


Figure 1: Diffractogram of pure crystalline solid

Amorphous solids under the x-ray beam behave in a similar manner. The difference being because every atom is arranged randomly, there are an infinite number of planes with their own spacing. As shown in figure 2, this correlates to a moderately low angle hump (called an amorphous hump) with all spacings that would correlate to higher angled peaks being nonexistent because of deconstructive interference.

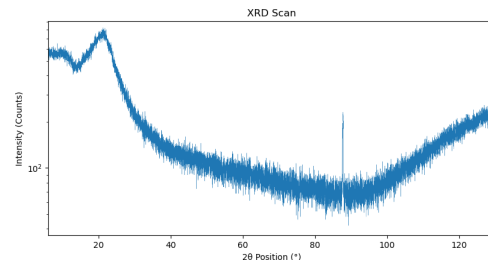


Figure 2: Diffractogram of pure amorphous solid

The materials analyzed in this work are silicon doped with phosphorous, silicon doped with boron, and pure silicon. The first two materials will be best thought of as a gradient from pure phosphorous or boron to pure silicon from top to bottom. Silicon is a crystalline material, while phosphorous and boron are amorphous. This means that the diffractogram will be a superposition of an amorphous diffractogram on a crystalline diffractogram. The silicon sample was only used to verify that the amorphous hump was present on the diffractograms.

The XRD process used to incrementally probe deeper depths is called grazing incidence x-ray diffraction (GIXRD, also called glancing incidence x-ray diffraction). Using this method the XRD will hold the incident beam at a fixed angle (called omega (ω)), while the diffracted beam side moves from a starting angle to an end angle. The detector is used in a 0D mode that simply counts all the x-rays that interact with it. The angle the counts correspond to is the angle the incident beam is added to the current diffracted beam angle all multiplied by 2: $2(\omega + \theta)$. After the scan finishes a new scan is started with a different incident angle to probe a different depth.

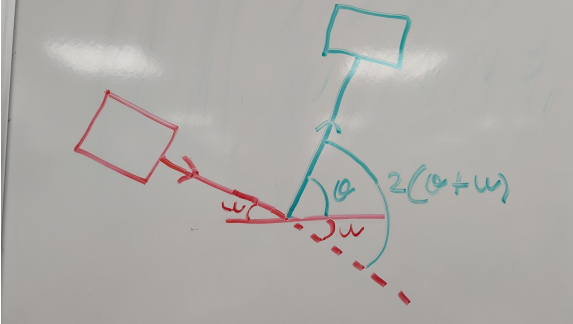


Figure 3: Diagram of XRD using GIXRD

The different incident angles probe different depths due to the different path lengths they provide the x-rays as they interact with the samples. This is due to the exponential attenuation of intensity as the beam travels given by this equation: $I(y) = I_0 e^{-\alpha y}$ with $\alpha = 2\omega n_I / c$. With the complex index of refraction (n_I) strongly dependent on wavelength. The x-ray source is copper, producing K_α , and K_β in most abundance. Since one wavelength provides the most uniformity on the diffractogram, the incident beam optics include a filter to limit the unnecessary wavelength. K_α is what is most common, so the K_β gets mostly filtered out by a nickel filter. This results in a x-ray beam that has a wavelength of 1.54\AA (uniformity in the wavelength is also necessary for Bragg's law).

2 Methods

Three different methods for determining relative percentages of the dopant in the substrate are used. The methods are: 1. Take the ratio of the max peak intensity, 2. Fit the peaks to a Gaussian curve, and find the integrated area, 3. Perform a summation to find the numerical integral. These methods are predicated on certain assumptions. The mixture of the probed depth is uniform, which is a known simplification of how the diffused dopants actually spread (an analysis to correct this has been tried, but better data is needed for it to be effective). The intensity of the x-ray is also assumed to have attenuated by 90%, and all relevant data comes from that. The last assumption is that the sample is put in its preferred orientation for the silicon structure to be aligned with the x-ray beam.

Method 1: Ratio

With this method the program takes the max intensity of the amorphous hump, and the max intensity of the most intense crystalline peak, and takes the ratio of them. This ratio is then the relative percentages of the dopant and substrate.

Method 2: Curve Fitting

This method takes the position and intensity values as the x and y inputs of scipy's curve_fit function. It then outputs the best fit for amplitude, mean, standard deviation, and a background value. The function then gets integrated, and the ratio of the areas are taken.

Method 3: Summation

This is an integral method as in method 2, but instead of fitting the data to a curve the data is integrated using discrete methods. The dx is the same for all points, so it is ignored with the foresight that it will be divided out. The sum is of all the intensity values for the peak or hump. A ratio is then taken from these sums.

3 Results

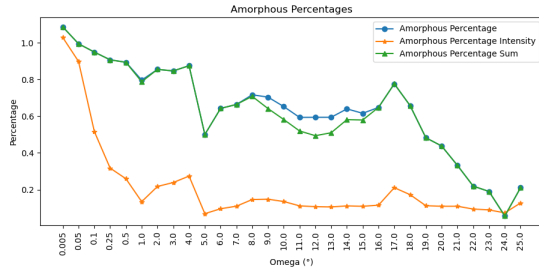


Figure 4: Percent amorphous at selected omegas

Figure 4 provides a concise overview of how the dopant and substrate mixture change with omega. All three methods show the amorphous percent decreases with depth, which is to be expected. They also show that the 0.005 omega is greater than 1 for the amorphous. This is due to the crystalline peak being nonexistent, but the program is still trying to see a peak. The curve fitting and summation method provide excellent agreement with each other, while the ratio method alludes to an exponential.

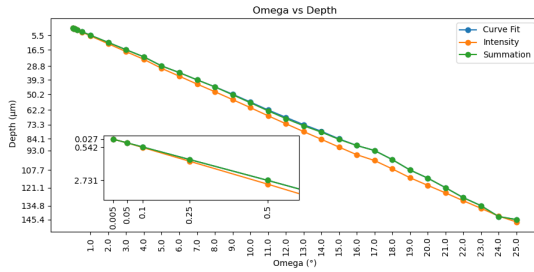


Figure 5: Depth vs omega for the three methods

Figure 5 shows the depth that correlates with the three methods. The methods have a close agreement, with the curve fit and summation method being right on top of each other. They all show a mostly linear relationship, which is to be expected since the indices of refraction are very similar for the dopant and substrate. Nevertheless, it is interesting that the methods more closely agree at lower and high omegas, but not the middle omegas.

4 Conclusion

Using GIXRD to determine relative composition of samples led to three analysis methods. Two of the methods utilized integration, and their results are closer aligned. One method used peak intensity ratios, and provides results that noticeably differ. All three methods provide reasonable agreement as to the depth that each omega probed. Using GIXRD to see relative mixture amount dependent on depth has been shown to be useful, keeping in mind the limitations given by the assumptions.

5 Future Work

Performing this analysis on a known sample would be able to provide greater validity to the results of this analysis. Creating an analysis to correct for the real diffusion process by not assuming uniform layers. This would produce the greatest impact on the real world application of these results.