

## Feature Article

## Transferring lead-free piezoelectric ceramics into application

Jürgen Rödel<sup>a,\*</sup>, Kyle G. Webber<sup>a</sup>, Robert Dittmer<sup>a</sup>, Wook Jo<sup>b</sup>, Masahiko Kimura<sup>c</sup>,  
Dragan Damjanovic<sup>d</sup><sup>a</sup> Institute of Materials Science, Technische Universität Darmstadt, Darmstadt, Germany<sup>b</sup> School of Mechanical and Advanced Materials Engineering, UNIST, Ulsan, Republic of Korea<sup>c</sup> Materials Technology Center, Murata Manufacturing Company Ltd, Japan<sup>d</sup> Ceramics Laboratory, Institute of Materials, School of Engineering, EPFL, Lausanne, Switzerland

Received 30 November 2014; accepted 11 December 2014

Available online 9 January 2015

## Abstract

After twenty years of partly quiet and ten years of partly enthusiastic research into lead-free piezoceramics there are now clear prospects for transfer into applications in some areas. This mimics prior research into eliminating lead from other technologies that resulted in restricted lead use in batteries and dwindling use in other applications. A figure of merit analysis for key devices is presented and used to contrast lead-containing and lead-free piezoceramics. A number of existing applications emerge, where the usage of lead-free piezoceramics may be envisaged in the near future. A sufficient transition period to ensure reliability, however, is required. The use of lead-free piezoceramics for demanding applications with high reliability, displacements and frequency as well as a wide temperature range appears to remain in the distant future. New devices are outlined, where the figure of merit suggests skipping lead-containing piezoceramics altogether. Suggestions for the next pertinent research requirements are provided.

© 2014 Elsevier Ltd. All rights reserved.

**Keywords:** Lead-free ferroelectrics; Piezoelectrics; Piezoelectric transducer; Piezoelectric actuator; Ferroelectricity

## 1. Introduction

About ten years ago, a publication by Saito et al.,<sup>1</sup> fueled excitement and spurred wide-spread scientific activity in the quest to replace lead-containing piezoceramics represented by  $\text{Pb}(\text{Zr,Ti})\text{O}_3$  (PZT) with non-toxic alternatives. This publication came as a response to legislative activity by the European Union<sup>2–4</sup> geared to eliminate toxic substances from electrical and electronic equipment to reduce their impact on the environment and health. In the following years the number of annual refereed publications increased from about 60 in 2004 to about 400 in 2011.<sup>5</sup> The precursor of these more recent activities is probably a publication by Takenaka and colleagues in 1991.<sup>6</sup> The work of Saito et al.<sup>1</sup> got attention for identifying a mixture of morphotropic and polymorphic phase transition region

in a  $(\text{K,Na})\text{NbO}_3$ -based system (KNN) with PZT-like values of piezoelectric coefficients in textured ceramics.

We should mention here briefly that “lead-free piezoelectric materials” is a generic term that encompasses two general groups: one competes for the same applications as PZT and another excels in properties that are outside the range where PZT can be used. To the first group (in competition with PZT) belong KNN,  $(\text{Bi}_{0.5}\text{Na}_{0.5})\text{TiO}_3$ - $\text{BaTiO}_3$  (BNT-BT), and  $(\text{Ba,Ca})(\text{Zr,Ti})\text{O}_3$  (BCZT) based materials. The second group includes materials with properties with which PZT cannot compete. These materials are “inferior” to PZT in some sense but are superior in another. Examples include:  $\text{SiO}_2$ ,  $\text{AlN}$ ,  $\text{LiNbO}_3$  (single crystals), Aurivillius structures (Bi-based layered structure), and other high temperature piezoelectrics.<sup>7</sup>

Piezoceramics not only command a huge market (piezoactuators alone have in excess of 20 billion \$<sup>5</sup>), but also act as an enabling technology for other areas, such as microelectronics through positioning elements in photolithography, medical diagnostics through ultrasound imaging, sensors and actuators

\* Corresponding author. Tel.: +49 6151 16 6315.

E-mail address: [roedel@ceramics.tu-darmstadt.de](mailto:roedel@ceramics.tu-darmstadt.de) (J. Rödel).

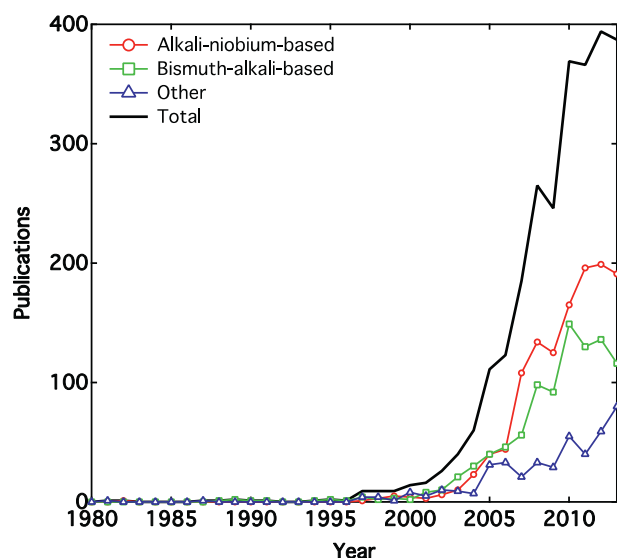


Fig. 1. Increase of the annual refereed publications during the last 34 years, divided into alkali-niobium-based and bismuth-alkali-based ceramics as well as others (mainly BCZT).

in automobile industry, and many others. Consequently, a series of review articles have appeared to aid in the international effort to develop new lead-free piezoelectric ceramics. These articles have typically attempted to survey the composition development process and, in addition, focused on various property issues that have arisen: the review by Damjanovic<sup>8</sup> was one of the first in the field and touched some specific domain wall-(micro)structure-properties relations, the paper by Shrout and Zhang<sup>9</sup> concentrated on a comparison to PZT, Takenaka et al.<sup>10</sup> considered mostly compositions based on BNT, while Li et al.<sup>11</sup> summarized the progress on KNN-based materials. Guidelines in terms of electronic structure, useful elements, and phase diagrams were the focus of the paper by Rödel et al.<sup>12</sup> five years ago. Other review efforts focused on the effects of dopants on electrical properties using a large number of tables to classify property changes,<sup>13</sup> while Aksel and Jones discussed crystallographic aspects.<sup>14</sup> The review by Coondoo et al.<sup>15</sup> provides an extensive dataset on lead-free piezoceramics, including bismuth layer structured ferroelectrics and langasites as well as a chapter on energy harvesting. Subgroups of lead-free piezoceramics are considered by Jo et al.,<sup>5</sup> in a description of incipient piezoelectrics and by Shvartsman and Lupascu with consideration of lead-free relaxors.<sup>16</sup> Recently, fatigue of lead-free ferroelectrics has been specifically addressed by Glaum and Hoffman.<sup>17</sup> Finally, a book on lead-free piezoceramics is currently available.<sup>18</sup> We, therefore, feel that the latest reviews have surveyed the field very effectively and see no need for a further assessment of scientific concepts or extensive discussion of the complete literature.

The latest literature survey, however, suggests that the scientific activity in this field has reached its peak, with about 400 refereed publications in 2010–2013 each (Fig. 1), indicating that the science in the field of lead-free piezoceramics has now reached a certain degree of maturity. In detail, research in KNN-based materials and BNT-based materials has saturated,

while new developments in BCZT bring up a third material opportunity. In the meantime, the computational community has developed high-throughput density functional theory, which could facilitate the prediction of experimentally discovered lead-free families that can compete with PZT.<sup>19</sup> Strong progress has been made with many compositions tested and an advanced, although still far from complete, understanding has now also been reached. The in-phase and out-of phase oxygen tilting, for example in the case of BNT may only serve as one example to illustrate the rich materials' complexity in lead-free piezoceramics.<sup>20</sup> With a certain degree of scientific maturity, more advanced methodologies like piezoforce microscopy,<sup>21</sup> in-situ transmission electron microscopy,<sup>22</sup> and in-situ diffraction techniques are considered.<sup>23,24</sup> In order to facilitate salient collaborations, research groups are expected to extend their level of international collaborations, to both academia and industry. With the fundamental understanding nearing, and in some cases even surpassing, the mechanistic understanding of lead-based ferroelectrics and ferroelectric relaxors, a societal responsibility toward application-oriented properties demands due attention. Hence, we have transgressed from an efficient, wide-spread scanning of a broad chemical parameter field toward both a detailed atomistic-based mechanistic understanding and a real-world demand-driven consideration of effects of temperature, frequency, stress, cycle number, processing issues, etc.

We suggest that the scientific and technological progress in the development of lead-free piezoceramics follows a generic picture with a peak of inflated expectations and a trough of disillusionment leading into a slope of enlightenment<sup>25</sup> (see Section 4). Given this assessment, let us, therefore, focus in this work on the transfer of our joint achievements into applications. In the main section we provide a classification of salient applications with respective figures of merit and contrast lead-containing piezoceramics with lead-free piezoceramics (Section 5). To illustrate our prospective path, we follow with select examples of already existing or expected industrial transfer into specific applications (Section 6). In the final sections we then suggest the next steps to take for fundamental as well as applied research (Section 7) and conclude with Section 8. We set the scene by reviewing issues of toxicity and ongoing legal matters (Section 2) and compare our activities to analogous strategies previously employed in different applications, where a replacement of lead has been achieved already (Section 3).

## 2. Toxicity of lead and legal issues

In the discussion about the toxicity of lead-containing ceramics such as PZT, it is important to differentiate the hazards involved in metallic lead, the precursor lead oxide and the final ceramic. The toxic effect of metallic lead is well known.<sup>26–28</sup> Due to the absorption from the lungs, skin, or the gastro-intestinal system, lead can be easily accumulated within the human body where it is stored in bone and soft tissue such as organs or muscles. The most immediate consequence is acute lead poisoning with severe damage to the blood as well as the gastro-intestinal and neurological systems.<sup>29</sup> The median lethal dose LD50, which is the dose sufficient to kill half of a tested

population, is reported to be 450 mg/kg, which is equivalent to 36 g of lead for a person with a mass of 80 kg.<sup>30</sup> Chronic lead exposure may result in hypertension, nephropathy, and, furthermore, affects the physical and mental development, especially of children.<sup>31,32</sup> Lead also impairs fertility, resulting in an increased probability of miscarriage, stillbirth, and neonatal deaths.<sup>33,34</sup>

In the production of lead-containing electroceramics, however, metallic lead is not used; only lead oxide PbO, which is less hazardous than Pb. The actual threshold limit value (TLV), which determines the highest air concentration of a hazardous substance that is not expected to result in acute or chronic damage, is 0.1 mg/cm<sup>3</sup> for PbO compared to 0.05 mg/cm<sup>3</sup> for Pb.<sup>35</sup> The LD50 then is 4300 mg/kg.<sup>36</sup> The toxicity of PbO is of particular concern as it has a high vapor pressure<sup>37</sup> and is fired at high temperatures.

It was demonstrated by Kosec et al.<sup>38</sup> that the final PZT ceramic may not be stable in an aqueous environment. Not only do electromechanical properties degrade in water but lead may also be dissolved and can be detected by means of ICP (induced coupled plasma) spectroscopy. This finding is especially alarming as a car may contain up to 100 g of PZT in small devices such as knock sensors, resistors, or piezoceramic injectors.<sup>39</sup> This is of particular concern because it may not be feasible to recycle all lead-containing components. In addition, due to the good etchability of PZT, acid rain also poses a serious concern in terms of lead leaching into the environment, e.g., groundwater. According to a publication by the United Nations Environment Programme, electronic waste generation volumes in developing countries exhibited an exponential growth in the past years.<sup>40</sup> The wide distribution of rather small components is critical for end-of-life vehicles as proper disposal or recycling is more complex than, for example, a lead acid battery. On a side note, some researchers see a similar issue with the heavy metal bismuth, which is used in many lead-free alternative piezoceramics. Its toxicity has been discussed in one of the review papers before,<sup>12</sup> but its use as the active ingredient bismuth subsalicylate in pepto-bismol,<sup>41</sup> an over-the-counter drug in the USA, may suffice as a further argument for its well-proven usage. A related issue concerns antimony, which is employed in a number of very promising lead-free piezoceramics. Here, one may note that antimony is also toxic and not suited to be part of a non-toxic replacement for PZT.

Consequently, legislation increasingly restricts the use of lead and lead-containing materials in order to reduce their impact on health and environment. With numerous regulations, the European Union (EU) is at the forefront of this movement, with others following.<sup>12</sup> For example, China has implemented a set of legislations by the Ministry of Industry and Information Technology of the People's Republic of China<sup>42</sup> with the following temporal evolution:

- I. Measures for Pollution Prevention and Management of Electronic Information Products Manufacturing, was issued in 2006 by Ministry of Industry and Information Technology of the People's Republic of China, and validated from March

- 1, 2007. Hazardous components, such as Pb, Cd, Cr<sup>6+</sup>, are restricted in electronic products.

- II. Regulations on Pollution Control of Electronic Information Products (draft for comment), was issued in June 2012 by Ministry of Industry and Information Technology of the People's Republic of China, National Development and Reform Commission, and Ministry of Environmental Protection of the People's Republic of China. Comments could be provided until July 10, 2012.

One of the most comprehensive set of rules by the EU is the REACH (Registration, Evaluation, Authorisation and Restriction of Chemical Substances) Regulation, which regulates the import and production of chemicals within the EU.<sup>43</sup> According to REACH, the fabrication, use, import, or sale of those hazardous substances, designated as substances of very high concern (SVHC), may eventually require authorization by the European Chemicals Agency (ECHA). As of June 2014, there are 155 SVHCs with PZT added in December 2012 due to its 'toxic for reproduction' classification.<sup>44</sup> SVHCs added to the Candidate List may subsequently be included in the Authorisation List (Annex XIV) of REACH, with the result that they are given a 'sunset date'. After this date, the substance must not be used or manufactured unless an exemption applies or authorization is provided for a specific use.

Another important regulation is the directive on the restriction of the use of certain hazardous substances in electrical and electronic equipment, abbreviated as RoHS, which was adopted in the European Union in 2003.<sup>45</sup> The most recent version of the regulation is the directive 2011/65/EU, which took effect on 2 January, 2013.<sup>46</sup> Consequently, the use of certain hazardous materials in electronic devices and electrical equipment is restricted within the EU. According to Annex 2 of the directive, this comprises lead, mercury, cadmium, hexavalent chromium, polybrominated biphenyl, and polybrominated diphenyl ethers. The regulation applies, among others, to consumer equipment, toys, medical devices, IT equipment, lighting equipment, and sports equipment. For example, lead in piezoceramics (appendix 7(c)-I) as well as in certain capacitors (7(c)-II and 7(c)-IV) is exempted because a substitution is currently technically impracticable. The validity of these exemptions expires after five years, that is, in July 2016, unless it is renewed.

However, the regulation does not apply to applications such as, for example, military equipment, spacebound devices, active implantable devices, photovoltaics, or R&D equipment. For a non-exempted application each homogeneous material within a certain device must contain less than 0.1% of the above-mentioned six substances. Then again, for certain cases exemptions are defined when a substitution is not possible for scientific and technical reasons, when reliability is not ensured, or when safety and health concerns and benefits from using a compound outweigh the benefit of a substitution (see Fig. 2). Environmental and health benefits of PZT-based devices can be significant in specific applications. Examples include less consumption of gasoline and cleaner air, enabled in part by piezoelectric fuel-injection actuators; noninvasive medical

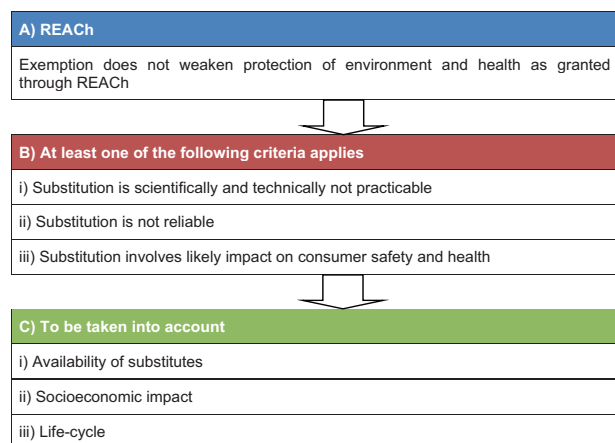
Fig. 2. Criteria for exemptions within RoHS.<sup>46</sup>

Table 1

List of exemptions in annex II of the directive on the restriction of the use of certain hazardous substances in electrical and electronic equipment as of November 2013. It is important to note that all cases are reviewed after the expiration date and a new expiration date may be established.

|          | Exemption  | Scope and Dates of Applicability  |
|----------|--|---|
| 7(c)-I   | Electrical and electronic components containing lead in a glass or ceramic other than dielectric ceramic in capacitors, e.g. piezoelectric devices, or in a glass or ceramic matrix compound | After 5 years (21 July 2016)  |
| 7(c)-II  | Lead in dielectric ceramic in capacitors for a rated voltage of 125 V AC or 250 V DC or higher   | After 5 years (21 July 2016)  |
| 7(c)-III | Lead in dielectric ceramic in capacitors for a rated voltage of less than 125 V AC or 250 V DC   | Expires on 1 January 2013 and after that date may be used in spare parts for EEE placed on the market before January 2013 |
| 7(c)-IV  | Lead in PZT based dielectric ceramic materials for capacitors which are part of integrated circuits or discrete semiconductors   | Expires on 21 July 2016   |

imaging and diagnosis, enabled by ultrasonic piezoelectric transducers; safety in transport, enabled by ultrasonic distance sensors; and structural health monitoring, enabled by piezoelectrically driven ultrasonic nondestructive testing. Specifically, health benefits from using PZT in ultrasonic medical imaging treatment and diagnosis far outweighs health hazard from unlikely uncontrolled disposal of a medical device in the environment at the end-of-life. The message is clearly that as long as an adequate alternative is not available, replacement of lead-based piezoelectrics used in critical applications such those listed should not be forced (Table 1).

Another regulation is the Directive on End-of-Life Vehicle 2000/53/EC, to be referred to as ELV in the following.<sup>47</sup> This

Table 2

List of exemptions in annex II of the Directive on End-of-Life Vehicles as of November 2013.

|       | Exemption   | Expiration date  |
|-------|---|--|
| 10(a) | Electrical and electronic components which contain lead in a glass or ceramic, in a glass or ceramic matrix compound, in a glass-ceramic material, or in a glass-ceramic matrix compound. |  |
| 10(b) | Lead in PZT based dielectric ceramic materials of capacitors being part of integrated circuits or discrete semiconductors   |  |
| 10(c) | Lead in dielectric ceramic materials of capacitors with a rated voltage of less than 125 V AC or 250 V DC   | Vehicle type approved before 1 January 2016 and spare parts for these vehicles |
| 10(d) | Lead in the dielectric ceramic materials of capacitors compensating the temperature-related deviations of sensors in ultrasonic sonar systems   | 2014   |

directive aims at reducing the amount of waste stemming from passenger cars and light commercial vehicles within the European Union. One of the main elements of ELV is pollution prevention by reducing the amount of hazardous substances in vehicles such as lead, mercury, cadmium, and hexavalent chromium. Again, certain exemptions are listed in the annex of the directive with revisions taking place regularly; the sixth revision amended the annex in May 2013.<sup>2</sup> Here, exemptions 10 (a–d) deal with lead in electric and electronic components as listed in Table 2.

Much of the worldwide R&D on lead-free piezoelectric materials in the last 10–15 years was triggered by the European legislations. Considering the number of applications that employ piezoelectric materials in industry, transport, and medicine, one would expect that the European Commission would follow its legislative actions by a concentrated support of the industrial, national, and academic research institutes working on developing new lead-free piezoelectric materials through its framework programs and now Horizon 2020. This has not happened. We could identify barely half a dozen projects financed by the Commission over the last 15 years with some activities on materials aspects of lead free piezoelectrics,<sup>48</sup> and only half of those were fully dedicated to materials research. In Japan, on the other hand, strong and sustained activities in development of new lead free piezoceramics were funded in the form of seven large research projects running over several years (2–5 years) and focusing on numerous materials, e.g., Ref. 49. In Korea, in 2013 two large projects started with a total financing of 13 million \$, which involves several companies, national laboratories, and universities. Besides the above team projects, based on search results using a database in National Science and Technology Information Service (<http://www.ntis.go.kr/>), 100 small sized projects on



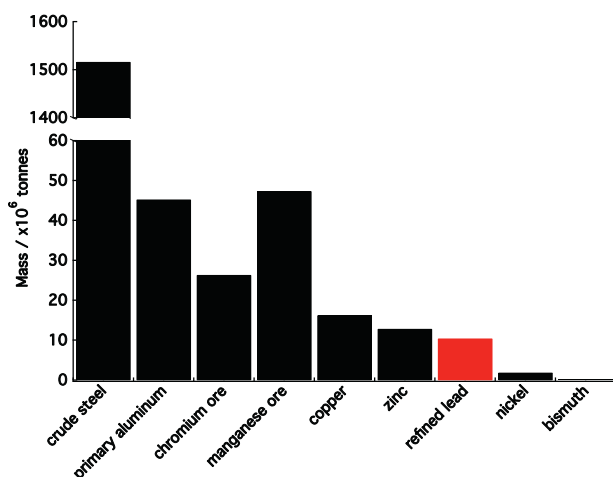


Fig. 3. World production of lead compared to other metals in 2011.<sup>50</sup>

lead-free piezoelectrics that have been carried out by individual researchers for the last decade were found that had funding in the range of US\$ 10–150 K per year. In China, however, the government-based funding agencies are the primary sponsors. There have been, for example, three key projects funded in 2014 by the National Natural Science Foundation (NSFC) on lead-free ferroelectrics.

### 3. World production of lead and its elimination outside of the battery world

According to the British Geological Survey,<sup>50</sup> the annual global production of refined metallic lead was 10.4 million tons in 2011, with 4.7 million tons stemming from actual mine production; the remainder came from recycling. By far the biggest producer is China with more than 4.6 million tons, followed by the USA (1.25 million tons), Germany (0.43 million tons), India (0.43 million tons), and South Korea (0.42 million tons). Therefore, lead is one of the most produced metals in the world with respect to weight as illustrated by Fig. 3.

Compared to 1970, the annual world production of refined lead has increased by 120% from 4.7 million tons, whereas mine production went up by almost 60% from 3.0 million tons.<sup>51</sup> Even in recent years the production of lead has significantly increased. From 2007 to 2011 the annual production of refined metallic lead has increased by 2.2 million tons or 27%, whereas the mine production went up by 1.0 million tons or 27%.<sup>50</sup> Compared to other metals such as aluminum, copper, or zinc, this evolution may appear like a comparably small increase (see Fig. 4). However, it appears astounding that the world production of lead is of comparable level to zinc and copper!

An increase of 27% is surprising when one considers that lead has been banished and replaced in a number of different applications. One of the most prominent examples for a replacement is the substitution of lead-containing additives in fuel. Used since the 1920s, tetraethyllead was a highly effective antiknock agent, increasing the efficiency and the peak power in engines. In 1972 the U.S. Environmental Protection Agency initiated the phasing out of leaded fuel, which began in 1975 and was fully

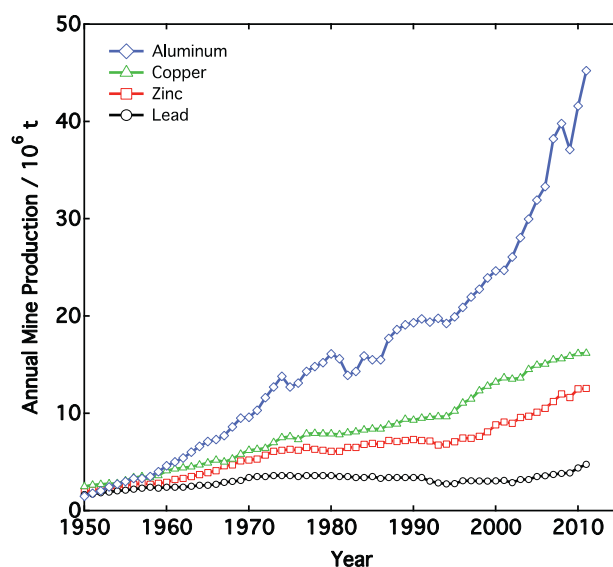


Fig. 4. Annual production of lead in megatons, compared to aluminum, copper, and zinc (according to Ref. 50).

accomplished in 1996.<sup>52,53</sup> In the European Union, restrictions on leaded fuel started in 1981 until it was eventually banned in 2000.<sup>4,54</sup> Similar regulations took effect almost everywhere in the world. It is most important to note that the economic benefits exceed the cost for the replacement of lead in fuels by a factor of ten. According to Tsai and Hatfield<sup>55</sup> the global annual benefit due to decreased morbidity and mortality adds up to more than 2 trillion US-\$. According to the United Nations, the only remaining countries in the world not selling unleaded fuel are Afghanistan, North-Korea, and Myanmar.<sup>56</sup>

Lead-free alternatives have also been developed for solders.<sup>57,58</sup> According to the U.S. Environmental Protection Agency about 80 thousand tons of lead-tin solder had been used worldwide in 2002.<sup>59</sup> However, owing to legislation lead-containing solders have been replaced to a great extent. A major driving force for that evolution has been RoHS, which effectively enforced lead-free solders within the European Union as of 1st of July 2006.<sup>45</sup> Furthermore, Japanese companies had been encouraged to use lead-free solders and China adopted its own regulations.<sup>60,61</sup> Moreover, the development of lead-free solders also became necessary as new applications like the hybrid electric car demanded higher working temperatures beyond 150 °C.<sup>62</sup> Conventional lead-containing solders such as Sn/37Pb, Sn62/Pb36/Ag2, or Sn63/Pb37 are not suited for temperatures that high.<sup>63</sup> Today, one of the most widely used solder is the ternary eutectic Sn/Ag/Cu, while other systems in use are mainly alloys of tin with copper, silver, or bismuth.<sup>57</sup>

Lead paint had been used since the days of the Roman Empire and the ancient Greeks owing to the good quality of white, yellow, and red pigments.<sup>64</sup> France, Belgium, and Austria banned white-lead interior paint as early as 1909,<sup>65</sup> but the worldwide movement of prohibiting lead paint only gained momentum in 1978 when the U.S. Consumer Product Safety Commission banned the use of lead paint in housing.<sup>66</sup> The consumption of lead in lead paints in the United States has consequently dropped from a peak value of more than 150 thousand tons per year to

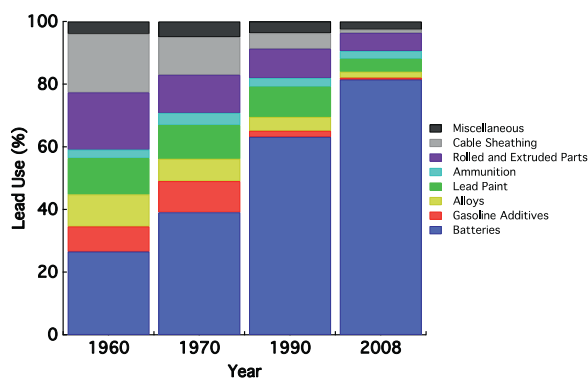


Fig. 5. End use of lead (after Refs. 76, 77).

nearly zero.<sup>67</sup> Furthermore, the European Union forbade the sale of lead paint beginning 1992.<sup>3</sup> While many more countries such as China, Brazil, and India adopted laws to restrict the use of lead paint, tests performed by NGOs indicate that the enforcement of these laws especially in developing countries is not always as comprehensive as desired.<sup>68</sup>

Other applications where lead has been replaced include ceramic glazes,<sup>69</sup> fishing sinkers,<sup>70,71</sup> food-processing equipment,<sup>72</sup> plumbing,<sup>73,74</sup> and lubricating greases.<sup>75</sup> As a consequence of these substitutions, the worldwide use of lead has largely changed since the 1960s. Today, more than 80% of all lead is used for the production of car batteries (Fig. 5). Thus, the increasing use of lead in the last 40 years has been due to an increased use of lead batteries. This knowledge should provide encouragement for the field of lead-free piezoceramics.

#### 4. Evolution of research

Let us first assess the achievements in research in a very broad manner by inspecting Fig. 1. EU legislation in combination with the 2004 paper by Saito et al.,<sup>1</sup> triggered research, leading to the steep rise in annual publications for the duration of time (to the year 2007) approximately needed for one Ph.D. thesis. These initial two triggers were followed by the investigations of Zhang et al.,<sup>78</sup> on BNT-BT-KNN and Liu and Ren<sup>79</sup> on BCZT. About two Ph.D. durations after the initial triggers (in 2011), a certain saturation has set in. From here on, the number of refereed publications per year has been nearly constant. Progress has been so rapid, that it has become difficult for new researchers to enter the field. The “obvious” innovations have been put forward, and from now on advancement requires deeper understanding, advanced methodology, and complementary expertise. Novel ideas can be imported by bringing researchers in from the consolidated field of lead-containing piezoceramics. If this does not happen, the research output in numbers of publications will go down. We have extended, therefore, the thick line (backed up by real data until 2013) by predictions using a dotted line from 2014 to 2030 (Fig. 6). To generalize the trend, we plot products (publications, parts, and devices) versus time in Fig. 6. The maximum in research productivity is often termed “peak of inflated expectations” with the following minimum being termed “trough of disillusionment”.<sup>25</sup>

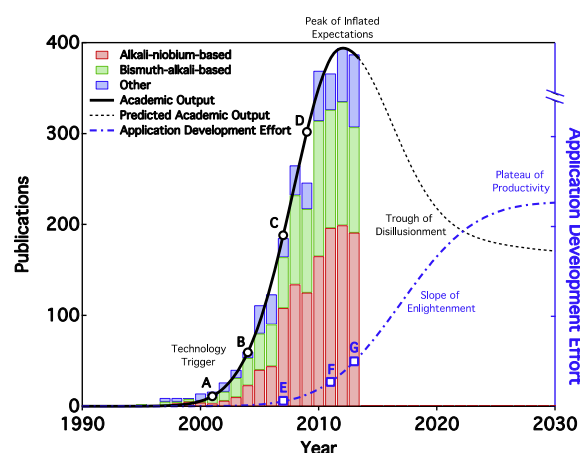


Fig. 6. Evolution of research (black) and transfer into application (blue) are contrasted in the time period from 1990 to 2030 with the evolution in research until 2013 stemming from the number of publications per year and the other curves predicted. Points indicate the following milestones: (A) Adoption of RoHS European Regulations, (B) Publication of Saito et al.,<sup>3</sup> (C) Publication of Zhang et al.,<sup>78</sup> (D) Publication of Liu and Ren,<sup>79</sup> (E) development of first commercial ultrasonic cleaner using lead-free piezoelectrics (Honda Electronics, BLT-type vibrator using BNT-BT), (F) development of lead-free multilayer actuator on water basis, and (G) medium-scale quantities (50 kg batches) for transducer applications such as flow meters.<sup>80</sup>

Academic arrogance may assume that solid research progress lures companies into the field, which may be driven to protect their pre-existing business based on lead-containing piezoceramics or because excellent, transferable expertise may enable new product lines challenging existing state-of-the-art technologies. In this latter case, however, several companies, like Toyota and Murata, e.g., Ref. 81, have been active in this field since the late 90s. Their expertise, in combination with new research funding in dynamic geographic areas and a newly created human potential eager to work in a field made attractive by its inherent nature of social responsibility, will determine the “slope of enlightenment”. If this picture holds true a plateau of productivity may be reached (in our prediction) between 2020 and 2025. Then, finally, the last sentence in the paper by Cross<sup>82</sup> may come true: “As the results (referring to Ref. 1) are brought to the market place, all producers may be forced to explore this option”. We need to critically examine in the next sections if this promise has potential to hold true and which applications are concerned as well as which applications are not ready and should not be currently required to go lead-free.

The decline in research output may seem as dreadful as the increase in applications seems unsubstantiated. As we are currently progressing from “peak of inflation” to “trough of disillusionment” it is timely to inquire how we can contribute to the technological “slope of enlightenment” and how we can substantiate our predictions for an increase in the number of applications.

We continue in the next section by classifying applications, defining figures of merit, and providing a direct comparison of the most advanced lead-free piezoceramics to lead-containing piezoceramics.

## 5. Competition between PZT and lead-free piezoceramics

### 5.1. Material parameters of concern

Piezoelectric materials are characterized by several physical parameters that bridge their electrical and mechanical properties. Due to the inherent crystallographic anisotropy of piezoelectric materials<sup>83</sup> as well as various types of available vibrational modes, the piezoelectric coefficient tensor is of the third rank and relates second rank mechanical and first rank electrical variables. During operation one of these variables denotes the input signal and the other denotes the output signal. The key examples of these properties include piezoelectric charge or strain coefficients ( $d_{ij}$ ), piezoelectric voltage or strain coefficients ( $g_{ij}$ ), electromechanical coupling coefficients ( $k_{ij}$ ), and mechanical quality factor ( $Q_m$ ), which are shown here in reduced notation or Voigt notation.<sup>83</sup> Tables 3 and 4 outline a number of applications and their respective figures of merit.

As the term ‘piezoelectricity’ suggests, one of the most important physical properties is the piezoelectric charge coefficient that correlates mechanical and electrical signals. When a small mechanical stress is applied to piezoelectric materials, an electrical displacement that is linearly proportional to the input mechanical force is generated and *vice versa*, an electric field generates strain. Since the electrical displacement requires charge accumulation at the electrodes, this proportionality is referred to as the piezoelectric charge coefficient or simply as piezoelectric coefficient. The same coefficient, now called strain coefficient, describes the converse effect. This piezoelectric charge or strain coefficient is known to be a consequence of the intrinsic piezoelectric effect and extrinsic reversible domain wall motion (also irreversible domain wall motion if the ac field is not small enough) in the case of ferroelectric-based piezoceramics,<sup>84</sup> and commonly called a small signal piezoelectric coefficient ( $d_{ij} = [pC/N] = [pm/V]$ ). When the input signal is large enough, e.g., larger than the local coercive field for domain walls, irreversible domain wall motion and partial switching contributions play a significant role, causing the piezoelectric coefficient to vary nonlinearly with the amplitude of the applied mechanical or electrical field. This dependence of piezoelectric coefficient on electric field and mechanical stress defines another useful parameter, the so-called large signal piezoelectric coefficient ( $d_{ij}^*$ ), which is especially useful in characterizing the actuating performance of piezoelectric materials. The large signal piezoelectric coefficient is simply given by the maximum achievable strain ( $S_{max}$ ) normalized by a given maximum electric field ( $S_{max}/E_{max}$ ), where  $E$  is usually unipolar. While in PZT small and large signal piezoelectric coefficient are empirically related, for example through a factor of 0.5,<sup>85</sup> the community working on lead-free piezoceramics has developed materials which exhibit a high large signal piezoelectric coefficient by purposely reducing the remanent polarization, which in turn diminishes the small signal piezoelectric coefficient.<sup>86</sup> This approach, however, can lead to a reduction in the usable temperature range and until recently, suffered from the requirement of high maximum electric fields in excess of 4 kV/mm.<sup>87</sup>

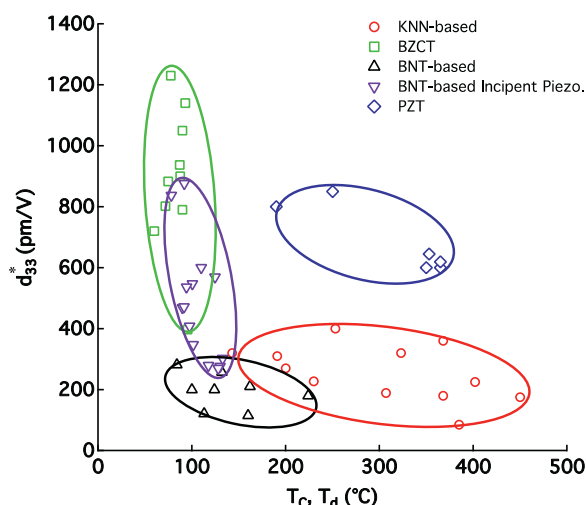


Fig. 7. The large signal piezoelectric coefficient (at room temperature) as a function of depolarization temperature for KNN-based<sup>1,9,88–90</sup> BCZT,<sup>79,91–94</sup> BNT-based,<sup>9,95–98</sup> BNT-based Incipient piezoelectrics,<sup>99–105</sup> and PZT.<sup>9,106,107</sup> It is important to note that the maximum applied electric field is not that same for each data point.

There are a number of key material parameters to be considered. In the following section we will outline the main figures of merit for some of the key applications and compare lead-free and lead-containing materials for each of these. In order to gain an overview of the various material systems, we have chosen to graphically represent two figures of merit as a function of Curie point or depolarization temperature: the large signal piezoelectric coefficient, important for actuator applications, and the small signal  $d \cdot g$  properties, used to evaluate materials for non-resonance sensing and energy harvesting applications. Key properties for actuator and sensor applications are large signal piezoelectric coefficient,  $d_{33}^*$ , and small signal piezoelectric coefficient,  $d_{33}$ , respectively. It has been shown before<sup>9,12</sup> that these are a strong function of the depolarization temperature ( $T_d$ ). Representative data of the large signal piezoelectric coefficient for lead-free and lead-containing piezoceramics are presented in Fig. 7. If several classes of piezoceramics are plotted, including lead-containing relaxors and barium titanate, a strong function of depolarization temperature is obtained. From Fig. 7 several features are apparent. High temperature applications with lead-free piezoceramics should be addressed with KNN-type materials. For actuators with usage temperatures below 100 °C, both (Ba,Ca)(Ti,Zr)O<sub>3</sub> and incipient piezoelectrics appear feasible candidates for transfer into application. In detail, one may note, that very soft lead-containing materials with very low depolarization temperature need to be considered as competitor for these low temperatures.

When the piezoelectric charge coefficient is divided by dielectric permittivity, ( $\epsilon_r \epsilon_0$ , where  $\epsilon_r$  and  $\epsilon_0$  are denoted as relative dielectric permittivity and permittivity of free space, respectively), another useful material parameter is derived. Given that charge divided by capacitance represents voltage, this parameter is called piezoelectric voltage coefficient ( $g_{ij}$ ), which measures the achievable voltage output for a given mechanical input signal, and alternatively, a mechanical

Table 3  
Current and future piezoelectric applications and their critical figures of merit (FOM).

| Temperature range                             | Resonant or non-resonant | Applications  |   | FOM                             | Other                    |
|---|--------------------------|---|---|---------------------------------|--------------------------|
|   |                          | Well-established  | Impending   |                                 |                          |
| Special use<br>$T > 500^\circ\text{C}$        | Resonant                 |   | Aerospace, Aircraft, nuclear power plant or geothermal power plant sensors <sup>127</sup>                   | $k^2 \cdot Q_m$                 |                          |
|   | Non-resonant             |   |   | $d \cdot g$                     |                          |
| SMD<br>$T > 250^\circ\text{C}$                | Resonant                 | Filter<br>Oscillator<br>Gyro sensor   |   | $k$<br>$Q_m$<br>$k^2 \cdot Q_m$ | $F_r - TC$<br>$F_r - TC$ |
|   | Non-resonant             | Acceleration sensor, HDD shock sensor   |   | $d \cdot g$                     |                          |
| Automotive<br>$T = 40$ to $180^\circ\text{C}$ | Resonant                 | Knocking sensor, Back sonar   | Energy harvesting (TPMS) <sup>128,129</sup>   | $k^2 \cdot Q_m$                 |                          |
|   | Non-resonant             | Knocking sensor   | Energy harvesting (TPMS) <sup>128,129</sup>   | $d \cdot g$                     |                          |
|   |                          | Fuel injection  |   | $S_{max}/E_{max}$               |                          |
|   | Resonant                 | Fish sonar, Flow meter, Medical probe   | Energy harvesting (Burglar alarm),<br>Ultrasonic transducer (data entry device),<br>Non destructive testing | $k^2 \cdot Q_m$                 |                          |
| Consumer<br>$T = -20$ to $80^\circ\text{C}$   |                          | Ultrasonic cleaner, Ultrasonic machining tool, <sup>a</sup> Camera lens autofocus (motor),<br>Power window (motor),<br>Backlight inverter,<br>High-voltage supply transformer | Wind blower, Air ionizer,   | $k^2 \cdot Q_m$                 | $v_{max}$                |
|   | Non-resonant             | Microphone  | Micro-mass sensor <sup>130</sup><br>Energy harvesting (Burglar alarm)                                       | $Q_m$<br>$d \cdot g$            |                          |
|   |                          | Stove burner, Lighter, Buzzer,<br>Vibration damping (sports equipment) <sup>131</sup>   |   | $d$                             |                          |
|   |                          | Ink jet, Loud speaker, Camera lens module   | HDD tracking, Pump,   | $S_{max}/E_{max}$               |                          |

<sup>a</sup> High power Langevin transducers are often used for ultrasonic machining tools. In this case large  $d_{33}$  values under a pre-stress are important.

displacement in response to accumulated charge density (also called strain coefficient). This parameter is especially useful in characterizing the sensitivity of sensors and the functionality of piezoelectric igniters. The piezoelectric strain coefficient  $d$  multiplied by the piezoelectric voltage coefficient  $g$  ( $d \cdot g$ ) is an important figure of merit for a number of non-resonant transduction applications, such as microphones and energy harvesting. Lead-free and lead-containing piezoceramics are contrasted with representative materials in Fig. 8. It is deduced that KNN becomes competitive for high-temperature applications and (Ba,Ca),(Ti,Zr)O<sub>3</sub> deserves attention for room temperature usage.

The square of the electromechanical coupling coefficient ( $k_{ij}$ ) represents the energy stored in a piezoelectric material when an electric field is applied while the mechanical distortion is constrained (note that it is a small signal property). This means that the electromechanical coupling coefficient is an indicator for conversion efficiency between electrical and mechanical energy. Ideally, the remaining (non-converted) applied energy is not lost.

Thus the word efficiency is not appropriate to describe coupling coefficients. It is also important to note that the electromechanical coupling factor only concerns the ratio between input and stored energy; thus, it does not represent the actual working efficiency of the material when utilized in transducer applications. Simply put, not all of the energy stored can be transferred to the mechanical output signal due to mechanical and electrical losses.<sup>123,124</sup> The normally quoted electromechanical coupling coefficients are the theoretical maximum values usually determined at the resonance frequency. When a piezoelectric material is exposed to an AC electric field with increasing frequency it starts to vibrate, and this vibration peaks at the mechanical resonance frequency with the absolute value of its impedance being the minimum (short circuit condition). This mechanical resonance frequency is determined by the geometry of the specimen along the observed vibration axis. This mechanical vibration dies away with further increasing driving frequency due to another resonance with electrical capacitance, which is commonly called ‘antiresonance’ (open circuit condition). The larger the distance



Table 4

Piezoelectric properties of lead-containing and lead free materials.

| $T_c$<br>°C | Application   | Material  | $T_c/T_d$<br>°C | $k_{33}$                             | $d_{33}$<br>pC/N or<br>pm/V | $g_{33}$<br>$10^{-3}$ Vm/N | $S_{max}/E_{max}$<br>pm/V | $Q_m$        | $k^2 \cdot Q_m$ | $d \cdot g$<br>pm <sup>3</sup> /J | $F_r - TC$<br>ppm/°C | $v_{max}$ | Ref |
|-------------|---|---|-----------------|--------------------------------------|-----------------------------|----------------------------|---------------------------|--------------|-----------------|-----------------------------------|----------------------|-----------|-----|
| >500        | Aerospace,<br>Aircraft and<br>Nuclear sensor, etc.                | BilnO <sub>3</sub> -PbTiO <sub>3</sub>                                    | 542             | 0.38( $k_t$ )                        | 60                          |                            |                           |              |                 |                                   |                      |           | 132 |
|             |   | La <sub>2</sub> Ti <sub>2</sub> O <sub>7</sub> (textured)                 | 1461            |                                      | 2.6                         |                            |                           |              |                 |                                   |                      |           | 133 |
|             |   | CaBi <sub>4</sub> Ti <sub>4</sub> O <sub>15</sub> (textured)              | 800             | 0.53( $k_t$ )                        | 45                          | 36.5                       |                           |              |                 | 1.6                               |                      |           | 134 |
|             |   | Na <sub>0.5</sub> Bi <sub>4.5</sub> Ti <sub>4</sub> O <sub>15</sub> based | 663             | 0.34( $k_t$ )                        | 30                          |                            |                           | 3800         | 440             |                                   |                      |           | 135 |
| >300        | Resonator   | PbTiO <sub>3</sub> based  | 310             | 0.35( $k_{15}$ )                     | 70( $d_{15}$ )              | 24                         |                           | 3000         | 368             | 1.7                               | 50                   |           |     |
|             |   | AT cut Quartz   | 576             | 0.11( $k_t$ )                        | 2                           | 50                         |                           | >50000       |                 |                                   | 0.1                  |           |     |
|             |   | SrBi <sub>2</sub> Nb <sub>2</sub> O <sub>9</sub> based<br>(textured)      | 350             | 0.21( $k_{15}$ )                     | 30( $d_{15}$ )              |                            |                           | 2850         | 126             |                                   | 0.4                  |           | 136 |
|             | Ultrasonic sensor <sup>a</sup> ,<br>motor                         | Hard Pb(Zr,Ti)O <sub>3</sub>  | 360             | 0.54<br>0.50( $k_{15}$ )             | 220<br>290( $d_{15}$ )      |                            |                           | 2000         | 583             |                                   |                      |           |     |
|             |   | (K,Na,Li)NbO <sub>3</sub> based   | 410             | 0.25( $k_{31}$ )<br>0.72( $k_{15}$ ) | 93<br>207( $d_{15}$ )       |                            |                           | 1400         | 726             |                                   |                      |           | 137 |
|             |   | (K,Na)(Nb,Ta)O <sub>3</sub> based   | 400(est.)       | 0.41( $k_p$ )                        |                             |                            |                           | 1400         |                 |                                   |                      |           | 138 |
|             |   | SrBi <sub>2</sub> Nb <sub>2</sub> O <sub>9</sub> based (textured)         | 350             | 0.33                                 | 27                          |                            |                           | 2320         | 253             |                                   |                      |           | 139 |
|             |   | Soft Pb(Zr,Ti)O <sub>3</sub>  | 300             | 0.71                                 | 410                         | 22                         | 600                       | 80           | 40              | 9.0                               |                      |           |     |
| >100        | Automotive sensor,<br>SMD sensor,<br>FI actuator                  | Soft Pb(Zr,Ti)O <sub>3</sub> (FI)   | 330             |                                      |                             |                            | 900                       |              |                 |                                   |                      |           |     |
|             |   | (K,Na)NbO <sub>3</sub> based  | 373             | 0.46( $k_p$ )                        | 262                         |                            |                           |              |                 |                                   |                      |           | 140 |
|             |   | (K,Na)NbO <sub>3</sub> based (textured)                                   | 323             |                                      | 373                         |                            | 672(est.)                 |              |                 |                                   |                      |           | 1   |
|             |   | Hard Pb(Zr,Ti)O <sub>3</sub>  | 360             | 0.54<br>0.50( $k_{15}$ )             | 220<br>290( $d_{15}$ )      |                            |                           | 2000<br>2000 | 583<br>500      |                                   |                      | 1         |     |
|             | Ultrasonic sensor<br>Ultrasonic cleaner,<br>Motor,<br>Transformer | (K,Na,Li)NbO <sub>3</sub> based   | 410             | 0.25( $k_{31}$ )<br>0.72( $k_{15}$ ) | 93<br>207( $d_{15}$ )       |                            |                           | 1400         | 726             |                                   |                      |           | 137 |



generation. In high power transducer applications, the performance often depends on the  $v_{\max}$  value.

The energy density in non-resonant sensors under the application of stress is proportional to both the piezoelectric charge ( $d_{ij}$ ) and voltage ( $g_{ij}$ ) coefficient, and the figure of merit is defined as being proportional to the product of these piezoelectric coefficients ( $d \cdot g$ ). Piezoelectric strain of non-resonant actuators is proportional to the piezoelectric  $d$  coefficient. However, the strain for actuators driven by high electric field is not proportional to the intrinsic  $d_{ij}$  coefficient because it strongly depends on the ferroelectric domain motion; in this case the strain is a nonlinear function of the field and is usually described by  $S_{\max}/E_{\max}$ . In the case of frequency control devices, such as piezoelectric filters, oscillators, or micro-mass sensors, the electromechanical coupling coefficient or mechanical quality factor is used as the main indicator for the suitability of the materials. Although characteristics of piezoelectric materials should be optimized for the device shape or peripheral circuits in actual use, electromechanical coupling coefficient, piezoelectric constants, mechanical quality factor, and their products are fundamentally set as figures of merit for many piezoelectric applications.

Piezoelectric characteristics of lead-containing and lead free materials are compared in Table 4 on the basis of the classification in Table 3. The characteristics denoted in blue cells were proposed as a figure of merit or important properties for these applications in Table 4. Some observations can be drawn from Table 4. In the intermediate and low temperature regime ultrasonic applications of KNN materials may become competitive with PZT-type materials. In the temperature regime with  $T < 100^\circ\text{C}$ , (Ba,Ca),(Ti,Zr)O<sub>3</sub> may become competitive for sensors and actuators, while incipient piezoelectrics based on BNT are attractive as actuators only, due to the relatively weak small signal piezoelectric coefficient.

Note, that the discussion above relies on only a subset of materials and properties identified as representative. In addition, there are a number of secondary properties that are essential for reliable and cost-effective manufacture (in addition to the financial cost of base materials) as well as for device operation, which are not directly related to electro-mechanical inter-conversion coefficients. We will consider these in turn.

### 5.2.1. Temperature dependence

During operation the temperature of a piezoelectric pressure sensor may change. A piezoelectric material that belongs to one of ten polar symmetries (which includes all piezoelectric ceramics) develops pyroelectric charge due to temperature sensitivity of the polarization.<sup>148</sup> In the case of ferroelectric materials, non-piezoelectric charge may also originate from thermally induced motion of domain walls. The generated noise may substantially degrade the performance of the sensor. Furthermore, elastic, dielectric and piezoelectric coefficients of piezoelectric materials are functions of temperature and may change appreciably in the vicinity of phase transitions. On the other hand, the higher Curie temperature of KNN (especially Li-doped KNN) may be turned into an advantage when compared to PZT.<sup>11</sup> Here, however, the phase transitions between ferroelectric phases may be

of more concern than the Curie temperature. This is true for KNN and BCZT based materials in which a so-called morphotropic phase boundary has a strong polymorphic character. This degree of the polymorphic character has recently been reduced and opened a new opportunity for tailoring KNN.<sup>141,149</sup> One of the notable advantages of quartz over PZT for sensor applications is exactly the remarkable thermal stability of its elastic constants.

### 5.2.2. Cycle dependence

Actuators may be subjected to  $10^9$  or more electrical cycles and need to remain in a narrow performance window. Cycling is usually done under unipolar driving fields, but can be required at different temperatures and under mechanical preload.<sup>17</sup> Thermal cycling during operation is also an important consideration that has received limited attention. Cyclic reliability is particularly important for certain applications, where replacing electroactive components is not possible or requires considerable effort, such as trailing edge flaps on large wind turbines or biomedical devices that are implanted within the body.

### 5.2.3. Dielectric permittivity

The piezoelectric voltage coefficient  $g$  is in part determined by the permittivity. If multiplied with the piezoelectric strain coefficient  $d$ , the resulting product ( $d \cdot g$ ) is an important figure of merit for non-resonant applications like parking pilots and energy harvesting systems (see Table 3). The large dielectric permittivity is undesired for operation at high frequencies, because it may lead to a low electrical impedance  $\propto 1/\omega C$ , where  $C$  is capacitance) and poor electrical impedance matching of the transducer with other electrical devices.<sup>150</sup>

### 5.2.4. Blocking force

In many actuator applications, strain needs to be provided against a certain mechanical load. This is, for example, the case in multilayer actuators that are preloaded to avoid cracking at the edges of the partial electrodes or that operate against a spring load. In these situations, a critical figure of merit is the blocking force, defined as the maximum force that can be generated in the fully clamped state, i.e., displacement held at zero during electrical activation.<sup>151</sup> Here, the conversion efficiency, namely the amount of strain that can be effectively converted into force,<sup>94</sup> is highly important and is controlled by the elastic properties and the particular mechanism responsible for strain generation, i.e., domain wall motion, field-induced phase transition.

### 5.2.5. Density

In underwater and medical applications the acoustic impedance of the transducer ( $\propto \sqrt{\rho c}$ , where  $\rho$  is the density and  $c$  speed of sound) should be close to that of the object under investigation (acoustic impedance matching). This favors materials with a lower density and ceramic-polymer composites are materials of choice.<sup>152</sup> The resonant frequency ( $f_r$ ) is related to the size of the sample ( $l$ ), the density ( $\rho$ ), and the mechanical compliance ( $s$ ), so that ( $f_r \propto 1/l\sqrt{\rho s}$ ). All other parameters being the same, a material with a lower density is advantageous for high frequency ( $>20$  MHz) transducers because for a given frequency it is easier to fabricate thicker elements. In such cases

lead-free materials have an advantage over lead-based materials. For example, KNN has a density of  $4.5 \text{ g/cm}^3$  and BNT has a density of  $5.3 \text{ g/cm}^3$ , compared to PZT with a density of  $8.0 \text{ g/cm}^3$  (see Section 6).

#### 5.2.6. Reliability and processing window

Evaporation of alkali elements and/or of bismuth during processing, e.g., sintering, may pose a more significant problem than the evaporation of lead. Therefore, the processing windows, especially in KNN-derived materials<sup>11,148</sup> have been described to be very narrow and may entail reliability issues in cases where select properties are reported on few samples. Scaling production to the required batch sizes for industrial applications but may be difficult to achieve. If large-scale production of lead-free piezoceramics is considered, availability of raw powders with controlled purity and morphology needs to be ascertained. Especially, carbonates with high sensitivity to humidity warrant special precautions. Furthermore, the long-term interplay between piezoelectric, electrode, and passivation requires further study.

#### 5.2.7. Cost

Tantalum and niobium, which are considerably more expensive than the raw materials in PZT, are often, or in the case of niobium always, used in typical KNN compositions. This will not appeal to potential customers or manufacturers. An advantage for KNN may lie in the opportunity to use low-cost nickel as an electrode material, which is not possible for PZT.<sup>153</sup> In comparison, only copper electrodes are available for PZT multilayer actuators.

We will reconsider some of the primary and secondary issues stated above in terms of action items in Section 7.

### 6. Case Studies on examples where lead-free piezoceramics have specific merit and have entered or are about to enter the marketplace

It is useful to start this section by mentioning that the first practically used piezoelectric materials were lead-free: quartz (silicon dioxide –  $\text{SiO}_2$ ), Rochelle salt (sodium potassium tartrate tetrahydrate –  $\text{KNaC}_4\text{H}_4\text{O}_6 \cdot 4\text{H}_2\text{O}$ ), KDP (potassium dihydrogen phosphate –  $\text{KH}_2\text{PO}_4$ ), and BT.<sup>154</sup> Lead-based piezoelectrics, for the most part PZT, have become widely used since the 1960s. However, even during this subsequent half a century of dominance of PZT, lead-free piezoelectric materials were employed in many applications where properties of PZT were not adequate. Examples of these lead-free materials (grouped by usage) include quartz or  $\text{GaPO}_4$  (which is isomorphic to  $\text{SiO}_2$ ), bismuth titanate based layer structures (Aurivillius phases), the langasite family,  $\text{LiNbO}_3$ ,  $\text{AlN}$ , and polyvinylidene fluoride (PVDF). Quartz is commonly used as an oscillator in watches and in filters because the elastic properties (which define resonance frequency) of PZT are strong functions of temperature. Furthermore, quartz and gallium phosphate are also employed in pressure sensors. In this case, PZT easily depolarizes at high pressures, has high conductivity at low frequencies and elevated temperatures, and exhibits strong charge-force hysteresis.<sup>155–159</sup>

Bismuth titanate-based layered structures are used for high temperature applications since PZT exhibits a low  $T_C$  ( $350^\circ\text{C}$  at morphotropic phase boundary), depolarizes easily at elevated temperatures, and exhibits charge drifts associated with pyroelectric effect and domain wall motion.<sup>7,155,160,161</sup> Next, the langasite family,  $\text{LiNbO}_3$  and  $\text{AlN}$  are found in applications requiring high temperature and high frequencies, where PZT is at a disadvantage due to a low  $T_C$ , easy depolarization, and a low mechanical quality factor.<sup>127</sup> Polyvinylidene fluoride,  $(-\text{C}_2\text{H}_2\text{F}_2)_n-$  or PVDF) is employed in sensor applications where its enhanced impedance matching with gases, liquids and biological systems, mechanical flexibility, and possibility to produce large thin layers over large areas give a large benefit as compared to PZT.<sup>162</sup>

In this section, however, we are more interested in cases where lead-free materials are meant to *replace* PZT, that is, in the large number of situations where PZT exhibits excellent properties and needs to be replaced not for technical but for environmental reasons as controlled through future legislation. It turns out that developing materials that are competitive with PZT presents considerable scientific and engineering challenges.

Currently, there are many reports in the literature on prototypes of lead-free piezoelectric devices and materials being developed for specific applications. Here we discuss two recent examples: an ultrasonic imaging probe developed by a transducer company and a ceramic material for a broad range of applications that is now commercially available.

*A high frequency transducer* – Merits of potassium niobate,  $\text{KNbO}_3$  (KN), for high frequency transducers have been already discussed in the literature.<sup>163,164</sup> Briefly, among the most attractive features of KN for high frequency transducer applications are: the high thickness electro-mechanical coupling coefficient ( $k_t \approx 69\%$ ) along directions close to pseudo-cubic  $[001]_C$  off-polar axis, high ferroelectric-ferroelectric transition temperature ( $\approx 220^\circ\text{C}$ ), good temperature stability of  $k_t$ , and low density ( $4.6 \text{ g/cm}^3$  in KN versus  $\approx 8.0 \text{ g/cm}^3$  in PZT).<sup>165</sup> As the resonance frequency is inversely proportional to the density and the thickness of the resonator, a low density means that a high frequency transducer may be fabricated with a significantly larger thickness than a PZT transducer operating at the same frequency, or may operate at a higher frequency for the same thickness. This is a significant advantage for transducers operating above 20 MHz where thickness is well below  $100 \mu\text{m}$ . Recently, a French consortium lead by Vermon (Tours, France) reported a high-frequency linear-array transducer (30 MHz) for *in vivo* skin imaging, Fig. 9.<sup>166</sup>

The array is fabricated from a 1–3 composite consisting of a domain engineered  $[001]_C$ -oriented KN single crystal and a polymer. The performance of the probe is reported to be comparable to that of a device fabricated from PZT ceramic and operating under the same conditions at 20 MHz, but with a better image resolution and penetration depth than achieved with the PZT probe.

*Availability of lead-free materials* – An important aspect in the wider application of lead-free piezoelectric materials is their commercial availability. Development of a lead-free composition with excellent properties in a university laboratory or even



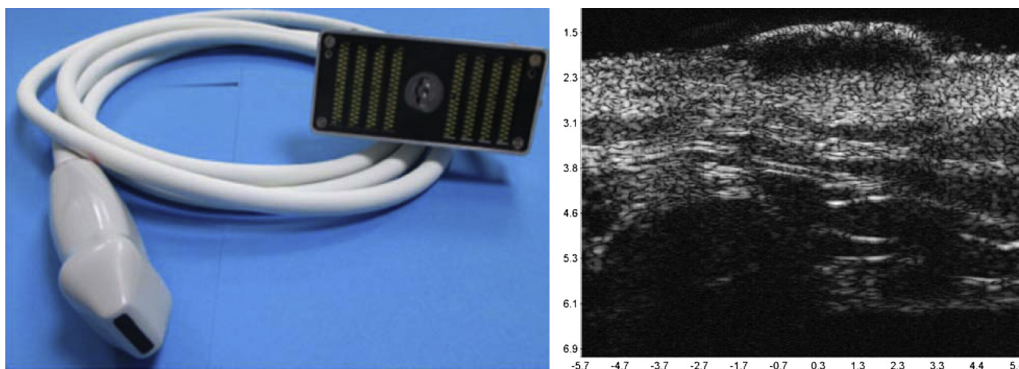


Fig. 9. Picture of a high-frequency linear-array transducer, which was used to image in vivo skin tissue. (Permission obtained from Vernon).

in an industrial research center does not bring much benefit to a wider community if that material cannot be produced reliably, in industrial quantities, and, most importantly, if it is not readily available to other users. This situation is now changing as shown by several material producers that offer various lead-free piezoelectric materials and devices, including lead-free based single crystals (KN and others by FEE<sup>167</sup>), lead-free transducers by PI<sup>80</sup> and ceramics (e.g., BNT-BT based by Honda Electronics<sup>168</sup>). The latter are advertised with the brand name “Lead Off” and are proposed for high power applications due to their excellent high mechanical quality factor up to high vibration velocity.

## 7. Steps to take for applied and basic research

We recall that PZT cannot be replaced by just one material.<sup>12</sup> Therefore, different applications are expected to see different material solutions that come with different challenges. The predicted line for technology transfer into applications in Fig. 6 then would be more accurately represented as a family of curves reaching the plateau of productivity in different years. As we are specifically interested in the first year of application of a specific product, it is expedient to transform Fig. 6 into Fig. 10 with the x-axis in years to realization.

Table 4 may now be utilized to extract different product realization timelines in an exemplary manner. We propose to use the following classification:

- (A) Lead-free piezoceramics with better properties than lead-containing piezoceramics.
- (B) Lead-containing piezoceramics can be replaced in ten years.
- (C) Lead-containing piezoceramics cannot be replaced in ten years.
- (D) New applications call for direct development of lead-free piezoceramics.

In the following, archetypical examples will be provided for each of the categories:

A: High power applications for ultrasonic motors feature a strongly decreasing mechanical quality factor with

increasing vibration velocity if manufactured from PZT. In contrast, both textured  $(\text{Sr,Ca})_2\text{NaNb}_5\text{O}_{15}$  as well as BNT-BT based materials maintain a high mechanical quality factor up to high vibration velocity and thereby can generate high output power density.<sup>169–171</sup> These are currently material solutions but not converted into device solutions as of yet. The aforementioned stringent reliability issues may necessitate a further development step of up to five years from now.

B: Ultrasonic cleaners are according to Table 3 in the category of consumer products with moderate temperature requirement. The figure of merit ( $k^2Q_m$ ) for lead-free piezoceramics is comparable to that for hard PZT. It is, therefore, not surprising that BNT-based piezoceramics have already been marketed for this application by Honda Electronics with relevant literature published several years ago.<sup>142</sup>

C: Actuators for fuel injection in automobiles require very high reliability, low cost, and need to provide high temperature stability and cycling stability with good normalized strain and

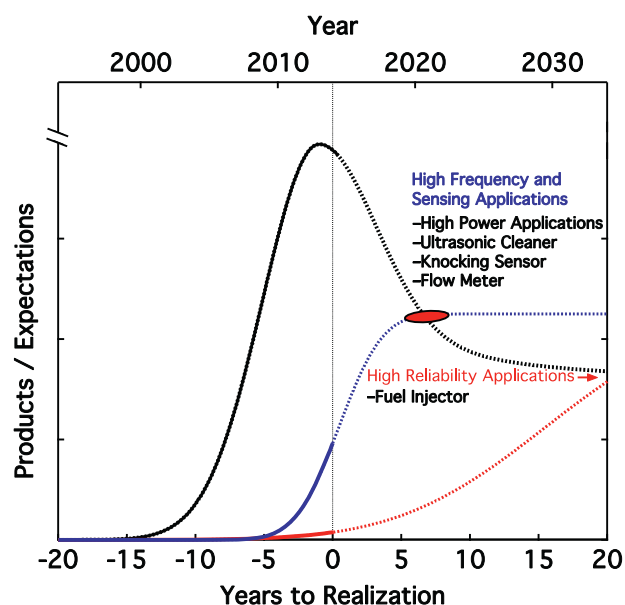


Fig. 10. Evolution of research and application transfer as a function of years to realization with some proposed applications earmarked and a time frame to realization predicted. (For interpretation of the references to color in the text, the reader is referred to the web version of the article.)

blocking force at high frequency.<sup>172</sup> This is, therefore, a component in a very competitive market, which sees very stringent multifold requirements and faces strict reliability requirements. The overall length of the piezo-injector is restricted as the free space under the engine hood is a safety requirement for crashes with pedestrians. The stringent reliability requirements are demonstrated by the fact that Continental Automotive GmbH had reached a total production of 50 million units in 2012. Hence, it is not suggested to replace this PZT component in the next ten years, as it also helps to provide energy-efficient engines with low emissions.

D: Some kinds of energy harvesting devices demanding a figure of merit of  $d \cdot g$  are competitive when manufactured in lead-free piezoceramics. In use, they are prone to be spread around the environment with little chance of being collected and recycled afterwards. These devices should go into production only in lead-free varieties. New applications with wide distribution of non-recyclable components that should go lead-free right from the start. A similar issue concerns flow and heat meters.

These products then can be positioned in Fig. 10. Materials in category A, B and D offer, therefore, feasible solutions if the necessary reliability requirements are met. These form a small percentage of all lead-free piezoceramic materials but offer a feasible start into the commercialization of these materials. They are on an advanced technology track and could offer product lines in about five to eight years from now (Points A, B, and D, elongated red dot). The overall development of these high frequency and sensing applications is schematically represented by the blue dotted line in Fig. 10. In contrast, several other applications, as exemplified by the piezoelectric fuel injector, do not yet offer a viable materials solution and do not provide a feasible date for transfer into application (red dotted line). It is expected that the development and, therefore, the number of years until commercialization will be longer than for smaller, less complex components, e.g., parking pilots that do not operate under high temperatures. This will result in a more gradual, adoption and product development slope, shown as the red dotted line in Fig. 10.

Next to the materials solution, there is also the issue of market pull in very dynamic and competitive market segments. Our inquiries resulted in the following predictions: In 2017, multilayer ceramic actuators for haptic applications may be used in mobile phones and in 2018, thick film piezoelectric speakers using lead-free ceramics for mobile phones. These, are, however, select opinions and not placed in our chart. Nevertheless they suffice to demonstrate the uncertainties related not only to materials but also to markets, scale, and competition.

Let us now discuss what remains to be done to move our worldwide scientific achievements into fruitful environmentally safe products. For the devices that can be marketed on a rather short-term basis, applied research recommendations are in due order. At the same time a number of fundamental research issues need to be addressed. We will consider both topics in turn. As we are concerned with a timely transfer of scientific achievements into application, we begin here with strongly warranted applied

research topics and then add few issues in the field of basic research.

## 7.1. Applied research

### 7.1.1. Temperature dependence

Structural issues determine the temperature dependence of lead-free piezoceramics. In KNN, optimum properties are obtained around the tetragonal to orthorhombic transition temperature,<sup>1,173–175</sup> where reduced properties are observed with an increasing temperature difference to the T-O phase transition temperature. For large piezoelectric coefficients, however, texturing can lead to good temperature insensitivity<sup>1</sup> as does decoupling the small signal and the large signal piezoelectric coefficients accomplished through tailoring the free energy barrier between two ferroelectric phases.<sup>88</sup> Similarly, for small-signal properties, like the coupling factor and the piezoelectric coefficient, good thermal stability has also been achieved using texturing, which naturally raises the question of financial viability.<sup>176</sup> As mentioned before, the development of KNN materials with MPB may provide a significant breakthrough. For BNT-based piezoceramics, this level of temperature insensitivity has not been reached and may, fundamentally, not be accessible due to the strong influence of the depolarization temperature on the remanent polarization and the large-signal piezoelectric coefficient.<sup>147</sup> BCZT, with its low Curie temperature, has a very limited temperature stability for piezoelectric properties.<sup>177</sup>

### 7.1.2. Frequency dependence

Frequency dependence of actuating strain in soft PZT is well described and provides a minor reduction of performance with increasing frequency.<sup>178</sup> Limited studies in KNN-based materials suggest that there is a strong compositional dependence with opportunities to obtain similar frequency dependence as in soft PZT.<sup>179</sup> In contrast, BNT-based materials can provide a very high strain through field-assisted phase change, but this mechanism suffers from a significant frequency dependence.<sup>180</sup>

### 7.1.3. Cycle dependence

The field has witnessed significant advancement in the understanding of electric fatigue for lead-free piezoceramics.<sup>17</sup> Salient application-oriented investigations are, however, still rare. In particular, influences of temperature<sup>181</sup> and uniaxial stress<sup>182</sup> are only available for PZT. Studies on the electric fatigue behavior of lead-free multilayer actuators are only now being performed and preferably would include issues of self-heating during electric loading.<sup>183</sup>

### 7.1.4. Blocking force

There are very limited studies available that directly characterize the blocking force of bulk ferroelectrics,<sup>151,184</sup> particularly in lead-free compositions.<sup>94,185</sup> Most previous studies focused on the displacement and force generation capabilities of a specific device that contain piezoelectric and non-piezoelectric components, such as bending actuators,<sup>186,187</sup> or functionally graded actuators with varying material properties.<sup>188</sup> These

applications often expose the electroactive material to a complex stress state that typically varies spatially.<sup>189</sup> Due to this, the obtained free displacement and blocking force values are not material parameters, but rather geometry-dependent values specific to a particular application. Recent work on bulk ferroelectrics has indicated that lead-free compositions are able to produce comparable, or even greater, blocking force values than commercial PZT, in some cases with 50% less free strain. The origin of the observations is not currently understood, although it is expected that the differences in displacement generation, i.e., piezoelectricity, ferroelectric domain wall motion, field-induced phase transitions, play an important role.

#### 7.1.5. Hardening and softening mechanisms:

Hardening through acceptor doping provides reduced dielectric losses and increased mechanical quality factor in PZT.<sup>126</sup> Although the effect has been known for a long time, there are still two competing models. These describe the orientation of an acceptor-oxygen vacancy complex or dimer,<sup>190</sup> in one case. In the other case, the charge drift of oxygen vacancies is considered.<sup>191</sup> Although applications for hard lead-free piezoceramics may require less stringent property requirements than for soft lead-free piezoceramics and appear to show great promise, this field has not yet attracted as many scientists as expected. However, doping with manganese in BNT-BT based materials<sup>192</sup> and in BNT-BKT-based materials<sup>193</sup> have yielded materials with increased mechanical quality factor  $Q_m$ . Enhanced mechanical quality factors have likewise been achieved in KNN-derived materials. This field is expected to grow strongly in the next years.<sup>194</sup> Similarly, softening mechanisms as described in soft PZT have not yet been detailed in lead-free piezoceramics.

#### 7.1.6. Mechanical properties

The fracture toughness of PZT is very close to the fracture toughness of window glass and thus very low.<sup>12</sup> An inherent toughening mechanism is provided through ferroelastic toughening,<sup>195</sup> while the most attractive composite toughening mechanism is transformation toughening.<sup>107</sup> The low fracture toughness is a very serious hindrance to device integration as multilayer actuators entail areas with severe strain incompatibility.<sup>196</sup> In particular, the interface to the electrode is prone to cracking<sup>197</sup> and will suffer from thermal expansion mismatch and electric field singularities.<sup>198</sup> Therefore, it is surprising that only very few studies are reported on lead-free piezoceramics. These include determinations of fracture toughness<sup>199</sup> and subcritical crack growth<sup>200</sup> in KNN-based materials, and fracture toughness in BNT.<sup>201</sup>

#### 7.1.7. Reproducibility

Poor reproducibility of the final functional properties remains among the major problems of lead-free compositions, preventing their wide use in practical applications.<sup>12,18,202</sup> The origins of the observed batch-to-batch variations can be in general ascribed to the following factors: hygroscopicity of the starting materials and/or some intermediate reaction products,<sup>203</sup> different diffusion rates resulting in local compositional inhomogeneities,<sup>204</sup>

high-temperature stability and volatilization,<sup>205,206</sup> specific sintering mechanisms,<sup>207</sup> and grain/domain-size dependence of the final properties.<sup>208,209</sup> Although these issues are repeatedly reported throughout the literature, only few systematic studies of processing conditions and mechanisms can be found. In order to assure reproducibility further knowledge of relevant chemical processes is needed, such as reliable phase diagrams, reaction kinetics, diffusion studies, detailed microstructural analyses, and others. At the same time, the adaptability of the synthesis processes to large-scale industrial production should be considered. This includes, among others, the use of inexpensive starting compounds of sufficient purity, aqueous processing routes, industrial-scale equipment, and optimized calcination and sintering conditions.

#### 7.1.8. Cost

Although RoHS considers the scientific and technological and socio-economic aspects, the price is not officially a criterion. However, replacing PZT is a tremendous endeavor and requires the careful consideration of companies' concerns, which naturally includes the overall production costs as well as competing technologies such as, e.g., magnetic fuel injection systems. For BNT-based materials, a phase evolution with Ag/Pd electrodes has been studied<sup>210</sup> and successful multilayers with these electrodes have been obtained via a water-based processing route.<sup>211</sup> Similarly, acceptable actuator performance has been obtained for KNN with Ag/Pd electrodes.<sup>212</sup> A breakthrough, which is estimated to be very important for the future, was obtained by manufacturing KNN with nickel inner electrodes.<sup>153</sup> The nickel inner electrodes in KNN have been shown by TEM to be structurally sharp without secondary phases.<sup>213</sup> The nickel is expected to provide low electro-migration and strong interfaces. As cofiring has to be performed under reducing atmosphere, electrical resistivity and dc degradation become an issue. Note, that the development of low cost solutions in this area has begun to attract the attention of large corporations (Murata, Taiyo Yuden) dominating the market of multilayer capacitors.<sup>153,213</sup>

#### 7.1.9. Electrical resistivity

In the development of base-metal capacitors, the conductivity as a function of oxygen partial pressure for doped barium titanate resulted in seminal studies by the group of D.M. Smyth in the early eighties.<sup>214</sup> This approach was adopted by numerous research groups, resulting in detailed electrical impedance studies of many materials.<sup>215</sup> Similarly, it is expected that the development of actuator materials with base-metal electrodes will require fundamental studies using impedance spectroscopy as a function of oxygen partial pressure. These investigations have only now begun for different gas atmospheres, for example in the case of BNT-BT materials.<sup>216</sup>

As befitting to the focus of this review, we propose some items for basic research. These are given less room here, but it is nevertheless clear, that focused fundamental research into lead-free piezoceramics will continue for many years and will be



fueled by the next successful transfers of non-toxic piezoceramics into applications, giving further emphasis to this research work. Before we conclude, we provide some suggestions for basic research topics.

## 7.2. Basic research

### 7.2.1. Atomistic structure

It appears that in particular the local structure in lead-free piezoceramics has remained somewhat elusive. This concerns the limited understanding to tailor lead-free piezoceramics using acceptor- and donor-doping and the relaxor character of BNT-based materials in particular. The interpretation and correlation to specific acceptor defect complexes in lead-containing ferroelectrics has reached now a mature level<sup>217</sup> if electron paramagnetic resonance (EPR) is contemplated. It has also benefitted strongly by the advances in density functional theory.<sup>218</sup> In contrast, however, the analysis of the EPR signal in acceptor-doped lead-free piezoceramics has just begun.<sup>219</sup> A very similar situation is observed if we consider nuclear magnetic resonance (NMR), which has been recently used to help elucidate the role of the local structure of lead-containing relaxors.<sup>220</sup> In contrast, application to lead-free piezoceramics has only rendered the very first results.<sup>221</sup> Funded coordinated programs with a combination of expertise in EPR, NMR, ab-initio computation, and possibly Raman spectroscopy<sup>222</sup> would, therefore, be highly beneficial.

### 7.2.2. ab-initio computations<sup>19</sup>

Atomistic computer simulations based on ab-initio methods such as density functional theory (DFT)<sup>223,224</sup> have been employed for a long time in the study of ferroelectric materials, especially since the introduction of the modern theory of polarization,<sup>225–227</sup> whose implementation into DFT programs allows calculation of spontaneous polarization and piezoelectric properties of any kind of material without input from experiment. Most studies focus on single-phase lead-free materials, such as KNN,<sup>228–230</sup> BNT,<sup>231–233</sup> or other Bi-based perovskites,<sup>234</sup> but also attempts to predict the morphotropic phase boundaries in lead-free solid solutions exist.<sup>231,235</sup> Another approach enabling prediction of so-far unknown materials allows the technique of high-throughput screening, which was applied to different perovskites<sup>19,236</sup> and other structures from crystallographic data bases<sup>237,238</sup> testing different elemental compositions in order to find new candidates with desirable physical properties and thermodynamic stability. The drawback of DFT calculation is their restriction to small system sizes up to 1000 atoms. It can be overcome by using DFT methods to develop effective Hamiltonians, which can be used in Monte Carlo simulations.<sup>239</sup> This allows the prediction of phase stabilities as a function of different variables such as composition, pressure, anisotropic strain, and temperature on larger length scales.<sup>240,241</sup>

### 7.2.3. Field dependent structure

Highest piezoelectric properties are obtained in the vicinity of phase transitions. These depend not only on temperature and composition, but also on electrical and mechanical fields,

particularly in the case of relaxor ferroelectrics.<sup>242</sup> This challenging field, particularly in the case of lead-free piezoelectrics, has initiated a wealth of innovative experimental techniques and may have surpassed the work on lead-containing piezoceramics with respect to novel in-depth understanding. BNT-based piezoceramics benefitted particularly from in-situ structural characterization, using synchrotron techniques<sup>243</sup> or a combination of synchrotron and neutron diffraction.<sup>244</sup> This material class was also deemed particularly intriguing to apply the very challenging in-situ transmission electron microscopy to render advanced understanding during increasing and decreasing electric fields.<sup>245,246</sup> More recently, very high strain BCZT materials were investigated using in-situ synchrotron diffraction.<sup>247</sup>

## 8. Conclusions

To summarize our findings in a few succinct bullet points:

- While legislation is not predictable, it is apparent that many other fields have successfully developed routes to lead-free solutions already. The predominant usage of lead is now restricted to lead-containing batteries, which can be easily recycled.
- Contrasting lead-free piezoceramics with lead-containing piezoceramics using salient figures of merit allows identification of different material classes
  - A: current applications, which are ready to go lead-free in the near future if critical issues of reliability are solved. This concerns, for example, some kinds of high power applications.
  - B: current applications, which can go lead-free in the foreseeable future (<10 years) if critical issues of reliability are solved. These are, for example, some kinds of ultrasonic transducers.
  - C: with applications, where a switch to lead-free piezoceramics may lie in the distant future (>10 years), for example piezo-injectors.
  - D: with new applications with wide distribution of non-recyclable components that should go lead-free right from the start. This could be materials for some kinds of flow meters and energy harvesting. The important caveat is that the needed piezoelectric materials are only partly commercially available.
- Application-oriented topics for the near future have been identified:
  - A: temperature-, field-, and cycle dependence.
  - B: blocking force and mechanical properties.
  - C: hardening, softening, and electrical resistivity.
  - D: most importantly: cost and reliability.
- Basics science issues are:
  - A: Atomistic structure in experiment and ab-initio computation
  - B: field-dependent investigations



## Acknowledgements

JR, RD and KGW at TUD are indebted to the Deutsche Forschungsgemeinschaft (DFG) for funding through SFB595/A1 and WE4972/2, respectively. DD acknowledges support of the FNS through PNR62 on lead-free materials No. 406240-126091. We are also thankful to a number of researchers who critically commented on an earlier version of this manuscript or provided helpful input: Jurij Koruza, Eric Patterson, and Melanie Gröting (TUD), Christian Hoffmann (TDK/EPCOS), Keisuke Kobayashi (Taiyo Yuden), Xiaoli Tan (Iowa State University), Klaus Reichmann (TU Graz), Jing-Feng Li and Ke Wang (Tsinghua University), Jae-Shin Lee (Ulsan University), Jérôme Acker (Continental Automotive GmbH), Michael Riedel (Piezoproducts), Bjørn Anderson (Noliac), Ruiping Wang (AIST), Daniel Rytz (FEE), Eberhard Hennig (PI Ceramics), Satoshi Wada (University of Yamaguchi), Julia Glaum (UNSW), Pim Groen (TNO), Gunnar Picht (Robert Bosch GmbH).

## References

- Saito Y, Takao H, Tani T, Nonoyama T, Takatori K, Homma T, et al. Lead-free piezoceramics. *Nature* 2004;**432**:84–7.
- EU-Commission Directive 2013/28/EU of 17 May 2013 amending annex II to directive 2000/53/EC of the European Parliament and of the Council on end-of-life vehicles. *Off J Eur Union* 2013;**L135**:14–8.
- EU-Council Directive 89/677/EEC of 21 December 1989 amending for the eighth time Directive 76/769/EEC on the approximation of the laws, regulations and administrative provisions of the member states relating to restrictions on the marketing and use of certain dangerous substances and preparations. *Off J Eur Commun* 1989;**L398**:19–23.
- EU-Commission Directive 2000/71/EC of 7 November 2000 to adapt the measuring methods as laid down in annexes I, II, III and IV to Directive 98/70/EC of the European Parliament and of the Council to technical progress as foreseen in article 10 of that directive. *Official Journal of the European Communities* 2000;**L287**:46–50.
- Jo W, Dittmer R, Acosta M, Zang J, Groh C, Sapper E, et al. Giant electric-field-induced strains in lead-free ceramics for actuator applications – status and perspective. *J Electroceram* 2012;**29**:71–93.
- Takenaka T, Maruyama K, Sakata K.  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ - $\text{BaTiO}_3$  system for lead-free piezoelectric ceramics. *Jpn J Appl Phys Part 1* 1991;**30**:2236–9.
- Turner RC, Fuierer PA, Newham RE, Shrout TR. Materials for high temperature acoustic and vibration sensors: a review. *Appl Acoust* 1994;**41**:299–324.
- Maeder MD, Damjanovic D, Setter N. Lead free piezoelectric materials. *J Electroceram* 2004;**13**:385–92.
- Shrout TR, Zhang SJ. Lead-free piezoelectric ceramics: alternatives for PZT? *J Electroceram* 2007;**19**:113–26.
- Takenaka T, Nagata H, Hiruma Y. Current developments and prospective of lead-free piezoelectric ceramics. *Jpn J Appl Phys* 2008;**47**:3787–801.
- Li J-F, Wang K, Zhu F-Y, Cheng L-Q, Yao F-Z.  $(\text{K}, \text{Na})\text{NbO}_3$ -based lead-free piezoceramics: fundamental aspects, processing technologies, and remaining challenges. *J Am Ceram Soc* 2013;**96**:3677–96.
- Rödel J, Jo W, Seifert KTP, Anton EM, Granzow T, Damjanovic D. Perspective on the development of lead-free piezoceramics. *J Am Ceram Soc* 2009;**92**:1153–77.
- Panda PK. Review: environmental friendly lead-free piezoelectric materials. *J Mater Sci* 2009;**44**:5049–62.
- Aksel E, Jones JL. Advances in lead-free piezoelectric materials for sensors and actuators. *Sensors* 2010;**10**:1935–54.
- Coondoo I, Panwar N, Kholkin A. Lead-free piezoelectrics: current status and perspectives. *J Adv Dielectr* 2013;**3**:1330002, 22 pp.
- Shvartsman VV, Lupascu DC. Lead-free relaxor ferroelectrics. *J Am Ceram Soc* 2012;**95**:1–26.
- Glaum J, Hoffman M. Electric fatigue of lead-free piezoelectric materials. *J Am Ceram Soc* 2014;**97**:665–80.
- Priya S, Nahm S, editors. *Lead-free piezoelectrics*. Netherlands: Springer; 2012.
- Armiento R, Kozinsky B, Fornari M, Ceder G. Screening for high-performance piezoelectrics using high-throughput density functional theory. *Phys Rev B* 2011;**84**:014103.
- Levin I, Reaney IM. Nano- and mesoscale structure of  $\text{Na}_{1/2}\text{Bi}_{1/2}\text{TiO}_3$ : a TEM perspective. *Adv Funct Mater* 2012;**22**:3445–52.
- Dittmer R, Jo W, Rödel J, Kalinin S, Balke N. Nanoscale insight into lead-free BNT-BT-xKNN. *Adv Funct Mater* 2012;**22**:4208–15.
- Guo H, Zhou C, Ren X, Tan X. Unique single-domain state in a polycrystalline ferroelectric ceramic. *Phys Rev B* 2014;**89**:014103.
- Jo W, Daniels JE, Jones JL, Xiaoli T, Thomas PA, Damjanovic D, et al. Evolving morphotropic phase boundary in lead-free  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ - $\text{BaTiO}_3$  piezoceramics. *J Appl Phys* 2011;**109**:014110.
- Ehmke MC, Glaum J, Hoffman M, Blendell JE, Bowman KJ. In situ X-ray diffraction of biased ferroelastic switching in tetragonal lead-free  $(1-x)\text{Ba}(\text{Zr}_{0.2}\text{Ti}_{0.8})\text{O}_3$ - $x(\text{Ba}_{0.7}\text{Ca}_{0.3})\text{TiO}_3$  piezoelectrics. *J Am Ceram Soc* 2013;**96**:2913–20.
- Fenn J, Raskino M. *Mastering the hype cycle: how to choose the right innovation at the right time*. Boston: Harvard Business Review Press; 2008.
- Goyer RA. Lead toxicity: current concerns. *Environ Health Perspect* 1993;**100**:177–87.
- Lockitch G. Perspectives on lead toxicity. *Clin Biochem* 1993;**26**:371–81.
- Patrick L. Lead toxicity, a review of the literature. Part 1: Exposure, evaluation, and treatment. *Anglais* 2006;**11**:2–22.
- Gordon JN, Taylor A, Bennett PN. Lead poisoning: case studies. *Br J Clin Pharmacol* 2002;**53**:451–8.
- Takahashi A. Problems of hygiene maintenance for food coming into contact with rubber and plastics products. *Nippon Gomu Kyokaishi* 1975;**48**:537.
- Lilis R, Gavrilescu N, Nestorescu B, Dumitriu C, Roventa A. Nephropathy in chronic lead poisoning. *Br J Ind Med* 1968;**25**:196–202.
- Perazella MA. Lead and the kidney: nephropathy, hypertension, and gout. *Conn Med* 1996;**60**:521–6.
- Rom WN. Effects of lead on reproduction. In: Infante PF, Legator MS, editors. *Proceedings of a workshop on methodology for assessing reproductive hazards in the workplace*. Washington, DC: National Institute for Occupational Safety and Health; 1980. p. 33–42.
- Apostoli P, Bellini A, Porru S, Bisanti L. The effect of lead on male fertility: a time to pregnancy (TTP) study. *Am J Ind Med* 2000;**38**:310–5.
- United States Department of Labor - Occupational Safety & Health Administration. *Occupational safety and health standard 1910.1025 App A: substance data sheet for occupational exposure to lead*; 1991.
- Canadian Centre for Occupational Health and Safety. *Material and safety data sheet on lead oxide*; 1995.
- Hardtl KH, Rau H. PbO vapour pressure in  $\text{Pb}(\text{Ti}_{1-x}\text{Zr}_x)\text{O}_3$  system. *Solid State Commun* 1969;**7**:41–5.
- Kosec M, Malic B, Wolny W, James A, Alemany C, Pardo L. Effect of a chemically aggressive environment on the electromechanical behaviour of modified lead titanate ceramics. *J Korean Phys Soc* 1998;**32**:S1163–S6.
- Information about Electrical Components which contain Lead in a Glass or Ceramic Matrix Compound. 2004. <http://ec.europa.eu/environment/waste/submissions/bosch3.pdf>
- Sustainable innovation and technology transfer industrial sector studies – recycling from e-Waste to resources*. United Nations Environmental Programme; 2009.
- Bierer DW. Bismuth subsalicylate: history, chemistry, and safety. *Clin Infect Dis* 1990;**12**:S3–8.
- <http://www.mii.gov.cn/n11293472/index.html>
- Regulation (EC) No 1907/2006 of the European Parliament and of the Council of 18 December 2006 concerning the Registration, Evaluation, Authorisation and Restriction of Chemicals (REACH), establishing a European Chemicals Agency, amending Directive 1999/45/EC and repealing Council Regulation (EEC) No 793/93 and Commission Regulation (EC)

- No 1488/94 as well as Council Directive 76/769/EEC and Commission Directives 91/155/EEC, 93/67/EEC, 93/105/EC and 2000/21/EC. *Official Journal of the European Union* 2006;**L396**:1–849.
44. European Chemicals Agency. *Inclusion of substances of very high concern in the candidate list*. vol. ED/169/2012; 2012.
  45. EU-Directive 2002/95/EC: restriction of the use of certain hazardous substances in electrical and electronic equipment (RoHS). *Off J Eur Union* 2003;**46**:19–37.
  46. EU-Directive 2011/65/EC: restriction of the use of certain hazardous substances in electrical and electronic equipment (RoHS). *Off J Eur Union* 2011;**54**:88–110.
  47. EU-Directive 2000/53/EC of the European Parliament and of the Council of 18 September 2000 on end-of life vehicles. *Off J Eur Union* 2000;**L269**:34–42.
  48. Kim JS, Wang KW, Smith EC. High-authority piezoelectric actuation system synthesis through mechanical resonance and electrical tailoring. *J Intell Mater Syst Struct* 2005;**16**:21–31.
  49. Yabuta H, Shimada M, Watanabe T, Hayashi J, Kubota M, Miura K, et al. Microstructure of  $\text{BaTiO}_3\text{--Bi}(\text{Mg}_{1/2}\text{Ti}_{1/2})\text{O}_3\text{--BiFeO}_3$  piezoelectric ceramics. *Jpn J Appl Phys* 2012;**51**:09LD4.
  50. World Mineral Statistics Electronic Archive: <https://http://www.bgs.ac.uk/mineralsuk/statistics/worldArchive.html>. British Geological Survey: Natural Environment Research Council.
  51. Armitage G. *Resources and energy statistics 2012*. Australian Government – Bureau of Resources and Energy Economics; 2013.
  52. United States Environmental Protection Agency. Regulation of fuels and fuel additives-notice of proposed rulemaking. *Fed Regist* 1972;**37**:3882–4.
  53. United States Environmental Protection Agency. Prohibition on gasoline containing lead or lead additives for highway use. *Fed Regist* 1996;**61**:3832–8.
  54. EU-Directive 98/70/EC of the European Parliament and of the Council of 13 October 1998 relating to the quality of petrol and diesel fuels and amending Council Directive 93/12/EEC. *Off J Eur Commun* 1998;**L350**:58–68.
  55. Peter LT, Hatfield TH. Global benefits from the phaseout of leaded fuel. *J Environ Health* 2011;**74**:8–14.
  56. United Nations Environment Programme. *Global clean fuels and vehicles database*; 2013.
  57. Abteu M, Selvaduray G. Lead-free solders in microelectronics. *Mater Sci Eng Rep* 2000;**27**:95–141.
  58. Suganuma K. Advances in lead-free electronics soldering. *Curr Opin Solid State Mater Sci* 2001;**5**:55–64.
  59. United States Environmental Protection Agency. *Solders in electronics: a life-cycle assessment*; 2005.
  60. Karl JP, Stalter KA. *Handbook of lead-free solder technology for micro-electronic assemblies*. New York City: Marcel Dekker, Inc.; 2004.
  61. Ministry of Commerce - People's Republic of China. *Measures for administration of the pollution control of electronic information products*; 2006.
  62. Renken F. High temperature electronics for future hybrid drive systems. In: *13th European Conference on Power Electronics and Applications (EPE)*, 2009. 2009. p. 1–7.
  63. Crusd A. Lead free solders in electronics. In: *Surface mount international conference proceedings*. 1998.
  64. Campos Gonzalez S. *Lead-based residential paint in soils: a dissolution and a spatial analysis prevention approach*, vol. 1458716. Ann Arbor: West Virginia University; 2008. p. 90.
  65. Kumar A, Gottesfeld P. Lead content in household paints in India. *Sci Total Environ* 2008;**407**:333–7.
  66. United States Consumer Product Safety Commission. *Ban of lead-containing paint and certain consumer products bearing lead-containing paint*. 16, vol. CFR 1303; 1977.
  67. Mielke H. Lead in the inner cities – policies to reduce children's exposure to lead may be overlooking a major source of lead in the environment. *Am Sci* 1999;**87**:62–73.
  68. Bodel R. *Lead legislation: the world's best and worst practice regulating lead in paint*. Sydney: Global Lead Advice & Support Service; 2010.
  69. Hinton JW. Lead-free glaze for alumina bodies. Google Patents; 1978.
  70. Minnesota Pollution Control Agency. Get the lead out!.
  71. Thomas VG. Attitudes and issues preventing bans on toxic lead shot and sinkers in North America and Europe. *Environmental Values* 1997;**6**:185–99.
  72. Smith GR. White copper-base alloy. Google Patents; 2012.
  73. La Fontaine A, Keast VJ. Compositional distributions in classical and lead-free brasses. *Mater Charact* 2006;**57**:424–9.
  74. Peters D. New bismuth/selenium red boss alloys solve lead concerns. *Mod Casting* 1997;**87**:57–9.
  75. Petro-Chem Industries. Extreme Pressure Grease (Lead Free).
  76. World Health Organization. *Inorganic lead (environmental health criteria)*. Geneva: World Health Organization; 1993.
  77. Taylor R, O'Brien E, Venkatesh T, Smith I. Lead Toxicity and Climate Change. In: *12th sustainable development conference*. 2009.
  78. Zhang ST, Kouna AB, Aulbach E, Ehrenberg H, Rödel J. Giant strain in lead-free piezoceramics  $\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3\text{--BaTiO}_3\text{--K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$  system. *Appl Phys Lett* 2007;**91**:112906.
  79. Liu WF, Ren XB. Large piezoelectric effect in Pb-free ceramics. *Phys Rev Lett* 2009;**103**:257602.
  80. Ditas P, Hennig E, Kynast A. *Lead-free piezoceramic materials for ultrasonic applications*. Berlin: VDE Verlag; 2014.
  81. Tani T. Crystalline-oriented piezoelectric bulk ceramics with a perovskite-type structure. *J Korean Phys Soc* 1998;**32**:S1217–20.
  82. Cross E. Materials science – lead-free at last. *Nature* 2004;**432**:24–5.
  83. Newnham RE. *Properties of materials: anisotropy, symmetry, structure*. Oxford: Oxford University Press; 2005.
  84. Pramanick A, Damjanovic D, Daniels JE, Nino JC, Jones JL. Origins of electro-mechanical coupling in polycrystalline ferroelectrics during sub-coercive electrical loading. *J Am Ceram Soc* 2011;**94**:293–309.
  85. Kerkamm I, Hiller P, Granzow T, Rödel J. Correlation of small- and large-signal properties of lead zirconate multilayer actuators. *Acta Mater* 2009;**57**:77–86.
  86. Jo W, Granzow T, Aulbach E, Rödel J, Damjanovic D. Origin of the large strain response in  $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$ -modified  $(\text{Bi}_{0.5}\text{Na}_{0.5})\text{TiO}_3\text{--BaTiO}_3$  lead-free piezoceramics. *J Appl Phys* 2009;**105**:094102.
  87. Groh C, Franzbach DJ, Jo W, Webber KG, Kling J, Schmitt LA, et al. Relaxor/ferroelectric composites: a solution in the quest for practically viable lead-free incipient piezoceramics. *Adv Funct Mater* 2014;**24**:356–62.
  88. Wang K, Yao F-Z, Jo W, Gobeljic D, Shvartsman VV, Lupascu DC, et al. Temperature-Insensitive  $(\text{K},\text{Na})\text{NbO}_3$ -based lead-free piezoactuator ceramics. *Adv Funct Mater* 2013;**23**:4079–86.
  89. Hollenstein E, Davis M, Damjanovic D, Setter N. Piezoelectric properties of Li- and Ta-modified  $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$  ceramics. *Appl Phys Lett* 2005;**87**:182905.
  90. Matsubara M, Yamaguchi T, Kikuta K, Hirano S. Sinterability and piezoelectric properties of  $(\text{K},\text{Na})\text{NbO}_3$  ceramics with novel sintering aid. *Jpn J Appl Phys* 2004;**43**:7159–63.
  91. Damjanovic D, Biancoli A, Batooli L, Vahabzadeh A, Trodahl J. Elastic, dielectric, and piezoelectric anomalies and Raman spectroscopy of  $0.5\text{Ba}(\text{Ti}_{0.8}\text{Zr}_{0.2})\text{O}_3\text{--}0.5(\text{Ba}_{0.7}\text{Ca}_{0.3})\text{TiO}_3$ . *Appl Phys Lett* 2012;**100**:192907.
  92. Ehmke MC, Ehrlich SN, Blendell JE, Bowman KJ. Phase coexistence and ferroelastic texture in high strain  $(1-x)\text{Ba}(\text{Zr}_{0.2}\text{Ti}_{0.8})\text{O}_3\text{--}x(\text{Ba}_{0.7}\text{Ca}_{0.3})\text{TiO}_3$ . *J Appl Phys* 2012;**111**:124110.
  93. Hao J, Bai W, Li W, Zhai J. Correlation between the microstructure and electrical properties in high-performance  $(\text{Ba}_{0.85}\text{Ca}_{0.15})(\text{Zr}_{0.1}\text{Ti}_{0.9})\text{O}_3$  lead-free piezoelectric ceramics. *J Am Ceram Soc* 2012;**95**:1998–2006.
  94. Brandt DRJ, Acosta M, Koruza J, Webber KG. Mechanical constitutive behavior and exceptional blocking force of lead-free BZT-xBCT piezoceramics. *J Appl Phys* 2014;**115**:204107.
  95. Dittmer R, Webber KG, Aulbach E, Jo W, Tan X, Rödel J. Electric-field-induced polarization and strain in  $0.94(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3\text{--}0.06\text{BaTiO}_3$  under uniaxial stress. *Acta Mater* 2013;**61**:1350–8.
  96. Anton E-M, Jo W, Damjanovic D, Rödel J. Determination of depolarization temperature of  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ -based lead-free piezoceramics. *J Appl Phys* 2011;**110**:094108.

97. Takenaka T, Nagata H. Current status and prospects of lead-free piezoelectric ceramics. *J Eur Ceram Soc* 2005;**25**:2693–700.
98. Zhang S-T, Kounga AB, Aulbach E, Deng Y. Temperature-dependent electrical properties of 0.94Bi<sub>0.5</sub>Na<sub>0.5</sub>TiO<sub>3</sub>-0.06BaTiO<sub>3</sub> ceramics. *J Am Ceram Soc* 2008;**91**:3950–4.
99. Hiruma Y, Imai Y, Watanabe Y, Nagata H, Takenaka T. Large electrostrain near the phase transition temperature of (Bi<sub>0.5</sub>Na<sub>0.5</sub>)TiO<sub>3</sub>-SrTiO<sub>3</sub> ferroelectric ceramics. *Appl Phys Lett* 2008;**92**:262904.
100. Hiruma Y, Nagata H, Takenaka T. Phase diagrams and electrical properties of (Bi<sub>1/2</sub>Na<sub>1/2</sub>)TiO<sub>3</sub>-based solid solutions. *J Appl Phys* 2008;**104**.
101. Zhang S-T, Kounga AB, Jo W, Jamin C, Seifert K, Granzow T, et al. High-strain lead-free antiferroelectric electrostrictors. *Adv Mater* 2009;**21**:4716–20.
102. Zhang ST, Kounga AB, Aulbach E, Granzow T, Jo W, Kleebe HJ, et al. Lead-free piezoceramics with giant strain in the system Bi<sub>0.5</sub>Na<sub>0.5</sub>TiO<sub>3</sub>-BaTiO<sub>3</sub>-K<sub>0.5</sub>Na<sub>0.5</sub>NbO<sub>3</sub>. I. Structure and room temperature properties. *J Appl Phys* 2008;**103**:034107.
103. Zhang ST, Kounga AB, Aulbach E, Jo W, Granzow T, Ehrenberg H, et al. Lead-free piezoceramics with giant strain in the system Bi<sub>0.5</sub>Na<sub>0.5</sub>TiO<sub>3</sub>-BaTiO<sub>3</sub>-K<sub>0.5</sub>Na<sub>0.5</sub>NbO<sub>3</sub>. II. Temperature dependent properties. *J Appl Phys* 2008;**103**:034108.
104. Acosta M, Jo W, Rödel J. Temperature and frequency dependent properties of the 0.75Bi<sub>1/2</sub>Na<sub>1/2</sub>TiO<sub>3</sub>-0.25SrTiO<sub>3</sub> lead-free incipient piezoceramic. *J Am Ceram Soc* 2014;**97**:1937–43.
105. Malik RA, Kang J-K, Hussain A, Ahn C-W, Han H-S, Lee J-S. High strain in lead-free Nb-doped Bi<sub>1/2</sub>(Na<sub>0.84</sub>K<sub>0.16</sub>)<sub>1/2</sub>TiO<sub>3</sub>-SrTiO<sub>3</sub> incipient piezoelectric ceramics. *Appl Phys Express* 2014;**7**:061502.
106. Kungl H, Theissmann R, Knapp M, Baecht C, Fuess H, Wagner S, et al. Estimation of strain from piezoelectric effect and domain switching in morphotropic PZT by combined analysis of macroscopic strain measurements and synchrotron X-ray data. *Acta Mater* 2007;**55**:1849–61.
107. Seo Y-H, Koruza J, Bencan A, Malic B, Rödel J, Webber KG. Simultaneous enhancement of fracture toughness and unipolar strain in Pb(Zr,Ti)O<sub>3</sub>-ZrO<sub>2</sub> composites through composition adjustment. *J Am Ceram Soc* 2014;**97**:1582–8.
108. Seo I-T, Choi C-H, Song D, Jang M-S, Kim B-Y, Nahm S, et al. Piezoelectric properties of lead-free piezoelectric ceramics and their energy harvester characteristics. *J Am Ceram Soc* 2013;**96**:1024–8.
109. Park H-Y, Seo I-T, Choi M-K, Nahm S, Lee H-G, Kang H-W, et al. Microstructure and piezoelectric properties of the CuO-added (Na<sub>0.5</sub>K<sub>0.5</sub>)(Nb<sub>0.97</sub>Sb<sub>0.03</sub>)O<sub>3</sub> lead-free piezoelectric ceramics. *J Appl Phys* 2008;**104**:034103.
110. Xue D, Zhou Y, Bao H, Zhou C, Gao J, Ren X. Elastic, piezoelectric, and dielectric properties of Ba(Zr<sub>0.2</sub>Ti<sub>0.8</sub>)O<sub>3</sub>-50(Ba<sub>0.7</sub>Ca<sub>0.3</sub>)TiO<sub>3</sub> Pb-free ceramic at the morphotropic phase boundary. *J Appl Phys* 2011;**109**:054110.
111. Tian Y, Chao X, Wei L, Liang P, Yang Z. Phase transition behavior and electrical properties of lead-free (Ba<sub>1-x</sub>Ca<sub>x</sub>)(Zr<sub>0.1</sub>Ti<sub>0.9</sub>)O<sub>3</sub> piezoelectric ceramics. *J Appl Phys* 2013;**113**:184107.
112. Taghaddos E, Hejazi M, Safari A. Electromechanical properties of acceptor-doped lead-free piezoelectric ceramics. *J Am Ceram Soc* 2014;**97**:1756–62.
113. Hiruma Y, Nagata H, Takenaka T. Depolarization temperature and piezoelectric properties of (Bi<sub>1/2</sub>Na<sub>1/2</sub>)TiO<sub>3</sub>-(Bi<sub>1/2</sub>Li<sub>1/2</sub>)TiO<sub>3</sub>-(Bi<sub>1/2</sub>K<sub>1/2</sub>)TiO<sub>3</sub> lead-free piezoelectric ceramics. *Ceram Int* 2009;**35**:117–20.
114. Wang XX, Tang XG, Chan HLW. Electromechanical and ferroelectric properties of (Bi<sub>1/2</sub>Na<sub>1/2</sub>)TiO<sub>3</sub>-(Bi<sub>1/2</sub>K<sub>1/2</sub>)TiO<sub>3</sub>-BaTiO<sub>3</sub> lead-free piezoelectric ceramics. *Appl Phys Lett* 2004;**85**:91.
115. Hiruma Y, Nagata H, Takenaka T. Thermal depoling process and piezoelectric properties of bismuth sodium titanate ceramics. *J Appl Phys* 2009;**105**:084112.
116. Choy SH, Wang XX, Chong CP, Chan HLW, Liu PCK, Choy CL. 0.90(Bi<sub>1/2</sub>Na<sub>1/2</sub>)TiO<sub>3</sub>-0.05(Bi<sub>1/2</sub>K<sub>1/2</sub>)TiO<sub>3</sub>-0.05BaTiO<sub>3</sub> transducer for ultrasonic wirebonding applications. *Appl Phys A* 2006;**84**:313–6.
117. Han H-S, Jo W, Kang J-K, Ahn C-W, Kim IW, Ahn K-K, et al. Incipient piezoelectrics and electrostriction behavior in Sn-doped Bi<sub>1/2</sub>(Na<sub>0.82</sub>K<sub>0.18</sub>)<sub>1/2</sub>TiO<sub>3</sub> lead-free ceramics. *J Appl Phys* 2013;**113**:154102.
118. Xu C, Lin D, Kwok KW. Structure, electrical properties and depolarization temperature of (Bi<sub>0.5</sub>Na<sub>0.5</sub>)TiO<sub>3</sub>-BaTiO<sub>3</sub> lead-free piezoelectric ceramics. *Solid State Sci* 2008;**10**:934–40.
119. Hiruma Y, Nagata H, Takenaka T. Detection of morphotropic phase boundary of (Bi<sub>1/2</sub>Na<sub>1/2</sub>)TiO<sub>3</sub>-Ba(Al<sub>1/2</sub>Sb<sub>1/2</sub>)O<sub>3</sub> solid-solution ceramics. *Appl Phys Lett* 2009;**95**:052903.
120. Ullah A, Ahn CW, Malik RA, Lee JS, Kim IW. Electromechanical and microstructural study of (1-x) Bi<sub>0.5</sub>(Na<sub>0.40</sub>K<sub>0.10</sub>)TiO<sub>3</sub>-x(Ba<sub>0.70</sub>Sr<sub>0.30</sub>)TiO<sub>3</sub> lead-free piezoelectric ceramics. *J Electroceram* 2014;**33**:187–94.
121. Fett T, Munz D, Thun G. Young's modulus of soft PZT from partial unloading test. *Ferroelectrics* 2002;**274**:67–81.
122. Piezoelectric Ceramic Products – Fundamentals, Characteristics and Applications. <http://piceramic.com/products/piezoelectric-materials.html>. PI Ceramics.
123. Uchino K, editor. *Piezoelectric actuators and ultrasonic motors*. Boston: Kluwer Academic Publishers; 1997.
124. Dittmer R, Webber KG, Aulbach E, Jo W, Tan X, Rödel J. Optimal working regime of lead-zirconate-titanate for actuation applications. *Sens Actuators A: Phys* 2013;**189**:187–94.
125. Senousy MS, Rajapakse R, Mumford D, Gadala MS. Self-heat generation in piezoelectric stack actuators used in fuel injectors. *Smart Mater Struct* 2009;**18**:045008.
126. Jaffe B, Cook WR, Jaffe H. *Piezoelectric ceramics*. London: Academic Press; 1971.
127. Zhang S, Yu F. Piezoelectric materials for high temperature sensors. *J Am Ceram Soc* 2011;**94**:3153–70.
128. Priya S. Advances in energy harvesting using low profile piezoelectric transducers. *J Electroceram* 2007;**19**:165–82.
129. Bedekar V, Oliver J, Priya S. Design and fabrication of bimorph transducer for optimal vibration energy harvesting. *IEEE Ultrason Freq Ferroelect Cntrl* 2010;**57**:1513–23.
130. Shin S, Song S, Lee Y, Lee N, Park J, Park H, et al. Fabrication and sensing behavior of piezoelectric microcantilever for nanobalance. *Jpn J Appl Phys* 2003;**42**:6139–42.
131. Viana FAC, Stedden Jr V. Multimodal vibration damping through piezoelectric patches and optimal resonant shunt circuits. *J Braz Soc Mech Sci Eng* 2006;**28**:293.
132. Zhang SJ, Xia R, Randall CA, Shrout TR, Duan RR, Speyer RF. Dielectric and piezoelectric properties of niobium-modified Bi<sub>1/2</sub>NO<sub>3</sub>-PbTiO<sub>3</sub> perovskite ceramics with high Curie temperatures. *J Mater Res* 2005;**20**:2067–71.
133. Yan H, Ning H, Kan Y, Wang P, Reece MJ. Piezoelectric ceramics with super-high Curie points. *J Am Ceram Soc* 2009;**92**:2270–5.
134. Takeuchi T, Tani T, Saito Y. Piezoelectric properties of bismuth layer-structured ferroelectric ceramics with a preferred orientation processed by the reactive templated grain growth method. *Jpn J Appl Phys* 1999;**38**:5553–6.
135. Wang C-M, Zhao L, Wang J-F, Zhang S, Shrout TR. Enhanced piezoelectric properties of sodium bismuth titanate (Na<sub>0.5</sub>Bi<sub>4.5</sub>Ti<sub>4</sub>O<sub>15</sub>) ceramics with B-site cobalt modification. *Phys Status Solidi RRL* 2009;**3**:7–9.
136. Kimura V, Ogawa H, Kuroda D, Sawada T, Higuchi Y, Takagi H, et al. Temperature dependence of piezoelectric properties for textured SrBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub> ceramics. *IEEE Trans Ultrason Ferroelectr Freq Control* 2007;**54**:2482–6.
137. Li E, Kakemoto H, Hoshina T, Tsurumi T. Shear-mode ultrasonic motor using potassium sodium niobate-based ceramics with high mechanical quality factor. *Jpn J Appl Phys* 2008;**47**:7702–6.
138. Matsubara M, Yamaguchi T, Sakamoto W, Kikuta K, Yogo T, Hirano S. Processing and piezoelectric properties of lead-free (K,Na)(Nb,Ta)O<sub>3</sub> ceramics. *J Am Ceram Soc* 2005;**88**:1190–6.
139. Kawada S, Ogawa H, Kimura M, Shiratsuyu K, Niimi H. High-power piezoelectric vibration characteristics of textured SrBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub> ceramics. *Jpn J Appl Phys* 2006;**45**:7455–9.
140. Wu J, Wang Y, Xiao D, Zhu J, Yu P, Wu L, et al. Piezoelectric properties of LiSBO<sub>3</sub>-Modified (K<sub>0.48</sub>Na<sub>0.52</sub>)NbO<sub>3</sub> lead-free ceramics. *Jpn J Appl Phys* 2007;**46**:7375–7.



141. Wang X, Wu J, Xiao D, Zhu J, Cheng X, Zheng T, et al. Giant piezoelectricity in potassium–sodium niobate lead-free ceramics. *J Am Chem Soc* 2014;**136**:2905–10.
142. Tou T, Hamaguti Y, Maida Y, Yamamori H, Takahashi K, Terashima Y. Properties of  $(\text{Bi}_{0.5}\text{Na}_{0.5})\text{TiO}_3$ – $\text{BaTiO}_3$ – $(\text{Bi}_{0.5}\text{Na}_{0.5})(\text{Mn}_{1/3}\text{Nb}_{2/3})\text{O}_3$  lead-free piezoelectric ceramics and its application to ultrasonic cleaner. *Jpn J Appl Phys* 2009;**48**:07GM3.
143. Shimizu H, Doshida Y, Mizuno Y, Tanaka S, Uematsu K, Tamura H. High-power piezoelectric characteristics of c-axis crystal-oriented  $(\text{Sr}, \text{Ca})_2\text{NaNb}_5\text{O}_{15}$  ceramics. *Jpn J Appl Phys* 2012;**51**:09LD02.
144. Shimizu H, Doshida Y, Tanaka S, Uematsu K. Piezoelectric properties of c-axis-oriented  $(\text{Sr}, \text{Ca})_2\text{NaNb}_5\text{O}_{15}$  piezoelectric ceramics with single-plate type and multilayered type fabricated using crystal-oriented sheet forming. *Key Eng Mater* 2010;**421**–**422**:21–5.
145. Tanaka D, Tsukada T, Furukawa M, Wada S, Kuroiwa Y. Thermal reliability of alkaline niobate-based lead-free piezoelectric ceramics. *Jpn J Appl Phys* 2009;**48**:09KD08.
146. Wang R, Bando H, Itoh M. *DO-033*. Xi'an, China: IMF-ISAF; 2009.
147. Wang K, Hussain A, Jo W, Rödel J. Temperature-dependent properties of  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ – $(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$ – $\text{SrTiO}_3$  lead-free piezoceramics. *J Am Ceram Soc* 2012;**95**:2241–7.
148. Lines ME, Glass AM. *Principles and applications of ferroelectrics and related materials*. Oxford: Clarendon Press; 1977.
149. Zushi J, Ariizumi T, Kojima S, Wang R, Bando H. Formation of morphotropic phase boundary in  $(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3$ – $\text{BaZrO}_3$ – $(\text{Bi}_{0.5}\text{Li}_{0.5})\text{TiO}_3$  lead-free piezoelectric ceramics. *Jpn J Appl Phys* 2013;**52**:07HB2.
150. Holterman J, Groen P. *An introduction to piezoelectric materials and components*. The Netherlands: Stichting Applied Piezo; 2012.
151. Webber KG, Franzbach DJ, Koruza J. Determination of the true operational range of a piezoelectric actuator. *J Am Ceram Soc* 2014;**97**:2842–9.
152. Gururaja TR, Schulze WA, Cross LE, Newnham RE, Auld BA, Wang YJ. Piezoelectric composite materials for ultrasonic transducer applications. Part I: Resonant modes of vibration of PZT rod-polymer composites. *IEEE Trans Son Ultrason* 1985;**32**:481–98.
153. Kawada S, Kimura M, Higuchi Y, Takagi H.  $(\text{K}, \text{Na})\text{NbO}_3$ -based multilayer piezoelectric ceramics with nickel inner electrodes. *Appl Phys Express* 2009;**2**:111401.
154. Cady WG. *Piezoelectricity*. New York: Dover Publications; 1964.
155. Berlincourt D, Krueger HHA, Jaffe B. Stability of phases in modified lead zirconate with variation in pressure, electric field, temperature and composition. *J Phys Chem Solids* 1964;**25**:659–74.
156. Morozov MI, Damjanovic D. Charge migration in  $\text{Pb}(\text{Zr}, \text{Ti})\text{O}_3$  ceramics and its relation to ageing, hardening and softening. *J Appl Phys* 2010;**107**:034106.
157. Krueger HHA, Berlincourt D. Effects of high static stress on the piezoelectric properties of transducer materials. *J Acoust Soc Am* 1961;**33**:1339–44.
158. Zhang QM, Pan WY, Jang SJ, Cross LE. Domain wall excitations and their contributions to the weak-signal response of doped lead zirconate titanate ceramics. *J Appl Phys* 1988;**64**:6445–51.
159. Krueger HHA. Stress sensitivity of piezoelectric ceramics: Part. 1. Sensitivity to compressive stress parallel to the polar axis. *J Acoust Soc Am* 1967;**42**:636–45.
160. Cook WR, Berlincourt DA, Scholz FJ. Thermal expansion and pyroelectricity in lead titanate zirconate and barium titanate. *J Appl Phys* 1963;**34**:1392–8.
161. Takenaka T, Sakata K, Toda K. Piezoelectric properties of bismuth layer-structured ferroelectric  $\text{Na}_0.5\text{Bi}_{4.5}\text{Ti}_4\text{O}_{15}$  ceramic. *Jpn J App Phys* 1985;**24**:730–2.
162. Vinogradov A, Holloway F. Electro-mechanical properties of the piezoelectric polymer PVDF. *Ferroelectrics* 1999;**226**:169–81.
163. Kari NM, Ritter TA, Park SE, Shrout TR, Shung KK. Investigation of potassium niobate as an ultrasonic transducer material. *Proc IEEE Ultrason Sym* 2000;**2**:1065–8.
164. Nakamura K, Kawamura Y. Orientation dependence of electromechanical coupling factors in  $\text{KNbO}_3$ . *IEEE Trans Ultrason Ferroelec Freq Control* 2000;**47**:750–5.
165. Davis M, Klein N, Damjanovic D, Setter N, Gross A, Wesemann V, et al. Large and stable thickness coupling coefficients of  $[001]_c$ -oriented  $\text{KNbO}_3$  and Li-modified  $(\text{K}, \text{Na})\text{NbO}_3$  single crystals. *Appl Phys Lett* 2007;**90**:062904.
166. Bantignies C, Filoux E, Mauchamp P, Dufait R, Pham Thi M, Rouffaud R, et al. Lead-free high-frequency linear-array transducer (30 MHz) for in vivo skin imaging. *IEEE Int Ultrason Sym* 2013:785–8.
167. <http://www.fee-io.de/>
168. Electronic H. *Ultrasonic transducer–piezoelectric ceramics/piezoelectric polymer film/piezoelectric thin film*; 2008 <http://www.honda-el.co.jp/ufile/file/329.pdf>
169. Hiruma Y, Watanabe T, Nagata H, Takenaka T. Piezoelectric properties of  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ -based solid solution for lead-free high-power applications. *Jpn J Appl Phys* 2008;**47**:7659–63.
170. Nagata H, Seki M, Noumura Y, Hiruma Y, Takenaka T. High-power piezoelectric characteristics of nontextured bismuth layer-structured ferroelectric ceramics. *Jpn J Appl Phys* 2011;**50**:09ND5.
171. Doshida Y, Shimizu H, Mizuno Y, Tamura H. Investigation of high-power properties of  $(\text{Bi}, \text{Na}, \text{Ba})\text{TiO}_3$  and  $(\text{Sr}, \text{Ca})_2\text{NaNb}_5\text{O}_{15}$  piezoelectric ceramics. *Jpn J Appl Phys* 2013;**52**:07HE01.
172. Randall CA, Kelnberger A, Yang GY, Eitel RE, Shrout TR. High strain piezoelectric multilayer actuators – a material science and engineering challenge. *J Electroceram* 2005;**14**:177–91.
173. Hollenstein E, Damjanovic D, Setter N. Temperature stability of the piezoelectric properties of Li-modified KNN ceramics. *J Eur Ceram Soc* 2007;**27**:4093–7.
174. Davis M, Klein N, Damjanovic D, Setter N, Gross A, Wesemann V, et al. Large and stable thickness coupling coefficients of  $[001]_c$  oriented  $\text{KNbO}_3$  and Li-modified  $(\text{K}, \text{Na})\text{NbO}_3$  single crystals. *Appl Phys Lett* 2007;**90**:062904.
175. Zhang S, Xia R, Hao H, Liu H, Shrout TR. Mitigation of thermal and fatigue behavior in  $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ -based lead free piezoceramics. *Appl Phys Lett* 2008;**92**:152904.
176. Chang Y, Poterala S, Yang Z, Messing GL. Enhanced electromechanical properties and temperature stability of textured  $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$ -based piezoelectric ceramics. *J Am Ceram Soc* 2011;**94**:2494–8.
177. Acosta M, Novak N, Jo W, Rödel J. Relationship between electromechanical properties and phase diagram in the  $\text{Ba}(\text{Zr}_{0.2}\text{Ti}_{0.8})\text{O}_{3-x}(\text{Ba}_{0.7}\text{Ca}_{0.3})\text{TiO}_3$  lead-free piezoceramic. *Acta Mater* 2014;**80**:48–55.
178. Masys AJ, Ren W, Yang G, Mukherjee BK. Piezoelectric strain in lead zirconate titanate ceramics as a function of electric field, frequency, and dc bias. *J Appl Phys* 2003;**94**:1155–62.
179. Zhou J-J, Wang K, Li F, Li J-F, Zhang X-W, Wang Q-M. High and frequency-insensitive converse piezoelectric coefficient obtained in  $\text{AgSbO}_3$ -modified  $(\text{Li}, \text{K}, \text{Na})(\text{Nb}, \text{Ta})\text{O}_3$  lead-free piezoceramics. *J Am Ceram Soc* 2013;**96**:519–23.
180. Dittmer R, Jo W, Aulbach E, Granzow T, Rödel J. Frequency-dependence of large-signal properties in lead-free piezoceramics. *J Appl Phys* 2013;**112**:014101.
181. Glaum J, Granzow T, Schmitt LA, Kleebe H-J, Rödel J. Temperature and driving field dependence of fatigue processes in PZT bulk ceramics. *Acta Mater* 2011;**59**:6083–92.
182. Wang H, Wereszczak AA, Lin H-T. Fatigue response of a PZT multilayer actuator under high-field electric cycling with mechanical preload. *J Appl Phys* 2009;**105**:014112.
183. Sapper E, Gassmann A, Gjodvad L, Jo W, Granzow T, Rödel J. Cycling stability of lead-free BNT-8BT and BNT-6BT-3KNN multilayer actuators and bulk ceramics. *J Eur Ceram Soc* 2014;**34**:653–61.
184. Webber KG, Aulbach E, Rödel J. High temperature blocking force measurements of soft lead zirconate titanate. *J Phys D Appl Phys* 2010;**43**:365401.
185. Dittmer R, Aulbach E, Jo W, Webber KG, Rödel J. Large blocking force in  $\text{Bi}_{1/2}\text{Na}_{1/2}\text{TiO}_3$ -based lead-free piezoceramics. *Ser Mater* 2012;**67**:100–3.
186. Haertling GH. Rainbow ceramics – a new type of ultra-high-displacement actuator. *Am Ceram Soc Bull* 1994;**73**:93–6.
187. Mulling J, Usher T, Dessent B, Palmer J, Franzon P, Grant E, et al. Load characterization of high displacement piezoelectric actuators with various end conditions. *Sens Actuators A* 2001;**94**:19–24.



188. Qiu JH, Tani J, Ueno T, Morita T, Takahashi H, Du HJ. Fabrication and high durability of functionally graded piezoelectric bending actuators. *Smart Mater Struct* 2003;**12**:115–21.
189. Woo S-C, Goo NS. Prediction of actuating displacement in a piezoelectric composite actuator with a thin sandwiched PZT plate by a finite element simulation. *J Mater Sci Technol* 2007;**21**:455–64.
190. Arlt G, Neumann H. Internal bias in ferroelectric ceramics – origin and time-dependence. *Ferroelectrics* 1988;**87**:109–20.
191. Genenko YA, Glaum J, Hirsch O, Kungl H, Hoffmann MJ, Granzow T. Aging of poled ferroelectric ceramics due to relaxation of random depolarization fields by space-charge accumulation near grain boundaries. *Phys Rev B* 2009;**80**:224109.
192. Zhu M, Liu L, Hou Y, Wang H, Yan H. Microstructure and electrical properties of MnO-doped  $(\text{Na}_{0.5}\text{Bi}_{0.5})_{0.92}\text{Ba}_{0.08}\text{TiO}_3$  lead-free piezoceramics. *J Am Ceram Soc* 2007;**90**:120–4.
193. Taghaddos E, Hejazi M, Safari A. Electromechanical properties of acceptor-doped lead-free piezoelectric ceramics. *J Am Ceram Soc* 2014;**96**:1756–62.
194. Li E, Kakemoto H, Wada S, Tsurumi T. Enhancement of  $Q_m$  by Co-doping of Li and Cu to potassium sodium niobate lead-free ceramics. *IEEE Trans Ultrason Ferroelectr Freq Control* 2008;**55**:980–7.
195. Schneider GA. Influence of electric field and mechanical stresses on the fracture of ferroelectrics. *Annu Rev Mater Res* 2007;**37**:491–538.
196. Kamlah M, Bohle U. Finite element analysis of piezoceramic components taking into account ferroelectric hysteresis behavior. *Int J Solids Struct* 2001;**38**:605–33.
197. Furuta A, Uchino K. Dynamic observation of crack-propagation in piezoelectric multilayer actuators. *J Am Ceram Soc* 1993;**76**:1615–7.
198. Lucato S, Lupascu DC, Kamlah M, Rödel J, Lynch CS. Constraint-induced crack initiation at electrode edges in piezoelectric ceramics. *Acta Mater* 2001;**49**:2751–9.
199. Yilmaz ED, Mgbemere HE, Oezcoban H, Fernandes RP, Schneider GA. Investigation of fracture toughness of modified  $(\text{K}_x\text{Na}_{1-x})\text{NbO}_3$  lead-free piezoelectric ceramics. *J Eur Ceram Soc* 2012;**32**:3339–44.
200. Zhang HJ, Li JX, Chu WY, Su YJ, Qiao LJ. Effect of humidity and hydrogen on the promotion of indentation crack growth in lead-free ferroelectric ceramics. *Mater Sci Eng B* 2010;**167**:147–52.
201. Jin DZ, Chen XM, Xu ZC. Influence of dispersed coarse grains on mechanical and piezoelectric properties in  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$  ceramics. *Mater Lett* 2004;**58**:1701–5.
202. Malič B, Benčan A, Rojac T, Kosec M. Lead-free piezoelectrics based on alkaline niobates: synthesis, sintering and microstructure. *Acta Chim Slov* 2008;**55**:719.
203. Hagh NM, Jadidian B, Safari A. Processing-property relationship in lead free KNN-solid solution system. *J Electroceram* 2007;**18**:339–46.
204. Malic B, Jenko D, Holc J, Hrovat M, Kosec M. Synthesis of sodium potassium niobate: a diffusion couples study. *J Am Ceram Soc* 2008;**91**:1916–22.
205. König J, Spreitzer M, Jancar B, Suvorov D, Samardzija Z, Popovic A. The thermal decomposition of  $\text{K}_{0.5}\text{Bi}_{0.5}\text{TiO}_3$  ceramics. *J Eur Ceram Soc* 2009;**29**:1695–701.
206. Popovič A, Bencze L, Koruza J, Malič B, Kosec M. Knudsen effusion mass spectrometric approach to the thermodynamics of  $\text{Na}_2\text{O-Nb}_2\text{O}_5$  system. *Int J Mass Spectrometry* 2012;**309**:70–8.
207. Koruza J, Malic B. Initial stage sintering mechanism of  $\text{NaNbO}_3$  and implications regarding the densification of alkaline niobates. *J Eur Ceram Soc* 2014;**34**:1971.
208. Hiruma Y, Nagata H, Takenaka T. Grain-size effect on electrical properties of  $(\text{Bi}_{1/2}\text{K}_{1/2})\text{TiO}_3$  ceramics. *Jpn J Appl Phys* 2007;**46**:1081–4.
209. Hao JG, Bai WF, Li W, Zhai JW. Correlation between the microstructure and electrical properties in high-performance  $(\text{Ba}_{0.85}\text{Ca}_{0.15})(\text{Zr}_{0.1}\text{Ti}_{0.9})\text{O}_3$  lead-free piezoelectric ceramics. *J Am Ceram Soc* 2012;**95**:1998–2006.
210. Schuetz D, Krauss W, Albering J, Kurta C, Reichmann K. The chemical interaction of silver-palladium alloy electrodes with bismuth-based piezomaterials. *J Am Ceram Soc* 2010;**93**:1142–7.
211. Krauss W, Schuetz D, Naderer M, Orosol D, Reichmann K. BNT-based multilayer device with large and temperature independent strain made by a water-based preparation process. *J Eur Ceram Soc* 2011;**31**:1857–60.
212. Kim M-S, Jeon S, Lee D-S, Jeong S-J, Song J-S. Lead-free NKN-5LT piezoelectric materials for multilayer ceramic actuator. *J Electroceram* 2009;**23**:372–5.
213. Kobayashi K, Doshida Y, Mizuno Y, Randall CA. Possibility of cofiring a nickel inner electrode in a  $(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3$ -LiF piezoelectric actuator. *Jpn J Appl Phys* 2013;**52**:09KD07.
214. Chan NH, Sharma RK, Smyth DM. Nonstoichiety in acceptor-doped  $\text{BaTiO}_3$ . *J Am Ceram Soc* 1982;**65**:167–70.
215. Chazono H, Kishi H. DC-electrical degradation of the BT-based material for multilayer ceramic capacitor with Ni internal electrode: impedance analysis and microstructure. *Jpn J Appl Phys* 2001;**40**:5624.
216. Zang J, Li M, Sinclair DC, Jo W, Rödel J. Impedance spectroscopy of  $(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ - $\text{BaTiO}_3$  ceramics modified with  $(\text{K}_{0.5}\text{Na}_{0.5})\text{NbO}_3$ . *J Am Ceram Soc* 2014;**97**:1523–9.
217. Eichel RA. *J Am Ceram Soc* 2008;**91**:691–701.
218. Eichel R-A, Erhart P, Träskelin P, Albe K, Kungl H, Hoffmann MJ. Defect-dipole formation in copper-doped  $\text{PbTiO}_3$  ferroelectrics. *Phys Rev Lett* 2008;**100**:095504.
219. Aksel E, Erdem E, Jakes P, Jones JL, Eichel RA. Defect structure and materials “hardening” in  $\text{Fe}_2\text{O}_3$ -doped  $(\text{Bi}_{0.5}\text{Na}_{0.5})\text{TiO}_3$  ferroelectrics. *Appl Phys Lett* 2010;**97**:012903.
220. Blinc R, Dolinsek J, Gregorovic A, Zalar B, Filipic C, Kutnjak Z, et al. NMR and the spherical random bond-random field model of relaxor ferroelectrics. *J Phys Chem Solids* 2000;**61**:177–83.
221. Aleksandrova IP, Ivanov YN, Sukhovski AA, Vakhrushev SB.  $^{23}\text{Na}$  NMR in the relaxor ferroelectric  $\text{Na}_{1/2}\text{Bi}_{1/2}\text{TiO}_3$ . *Phys Solid State* 2006;**48**:1120–3.
222. Aksel E, Forrester JS, Kowalski B, Deluca M, Damjanovic D, Jones JL. Structure and properties of Fe-modified  $\text{Na}_{0.5}\text{Bi}_{0.5}\text{TiO}_3$  at ambient and elevated temperature. *Phys Rev B* 2012;**85**:024121.
223. Hohenberg P, Kohn W. Inhomogeneous electron gas. *Phys Rev* 1964;**136**:B864–B71.
224. Kohn W, Sham LJ. Self-consistent equations including exchange and correlation effects. *Phys Rev* 1965;**140**:A1133.
225. King-Smith RD, Vanderbilt D. Theory of polarization of crystalline solids. *Phys Rev B* 1993;**47**:1651.
226. Resta R. Theory of the electric polarization in crystals. *Ferroelectrics* 1992;**136**:51–5.
227. Vanderbilt D, King-Smith RD. Electric polarization as a bulk quantity and its relation to surface-charge. *Phys Rev B* 1993;**48**:4442–55.
228. Körbel S, Marton P, Elsässer C. Formation of vacancies and copper substitutionals in potassium sodium niobate under various processing conditions. *Phys Rev B* 2010;**81**:174115.
229. Lu N, Yu R, Cheng Z, Dai Y, Zhang X, Zhu J. Ferroelectric polarization and domain walls in orthorhombic  $(\text{K}_{1-x}\text{Na}_x)\text{NbO}_3$  lead-free ferroelectric ceramics. *Appl Phys Lett* 2010;**96**:221905.
230. Suewattana M, Singh DJ. Local dynamics and structure of pure and Ta substituted  $(\text{K}_{1-x}\text{Na}_x)\text{NbO}_3$  from first principles calculations. *Phys Rev B* 2010;**82**:014114.
231. Gröting M, Albe K. Theoretical prediction of morphotropic compositions in  $\text{Na}_{1/2}\text{Bi}_{1/2}\text{TiO}_3$ -based solid solutions from transition pressures. *Phys Rev B* 2014;**89**:054105.
232. Gröting M, Hayn S, Albe K. Chemical order and local structure of the lead-free relaxor ferroelectric  $\text{Na}_{1/2}\text{Bi}_{1/2}\text{TiO}_3$ . *J Solid State Chem* 2011;**184**:2041–6.
233. Zeng M, Or SW, Chan HLW. First-principles study on the electronic and optical properties of  $\text{Na}_{0.5}\text{Bi}_{0.5}\text{TiO}_3$  lead-free piezoelectric crystal. *J Appl Phys* 2010;**107**:043513.
234. Baettig P, Schelle CF, LeSar R, Waghmare UV, Spaldin NA. Theoretical prediction of new high-performance lead-free piezoelectrics. *Chem Mater* 2005;**17**:1376–80.
235. Miura K, Furuta T. First-principles study of structural trend of  $\text{BiMO}_3$  and  $\text{BaMO}_3$ : relationship between tetragonal or rhombohedral structure and the tolerance factors. *Jpn J Appl Phys* 2010;**49**:031501.
236. Armiento R, Kozinsky B, Hautier G, Fornari M, Ceder G. High-throughput screening of perovskite alloys for piezoelectric performance and thermodynamic stability. *Phys Rev B* 2014;**89**:134103.

237. Bennett JW, Garrity KF, Rabe KM, Vanderbilt D. Hexagonal ABC semiconductors as ferroelectrics. *Phys Rev Lett* 2012;**109**:167602.
238. Bennett JW, Rabe KM. Integration of first-principles methods and crystallographic database searches for new ferroelectrics: strategies and explorations. *J Solid State Chem* 2012;**195**:21–31.
239. Zhong W, Vanderbilt D, Rabe KM. First-principles theory of ferroelectric phase-transitions for perovskites – the case of BaTiO<sub>3</sub>. *Phys Rev B* 1995;**52**:6301–12.
240. Akbarzadeh AR, Prosandeev S, Walter EJ, Al-Barakaty A, Bellaiche L. Finite-temperature properties of Ba(Zr,Ti)O<sub>3</sub> relaxors from first principles. *Phys Rev Lett* 2012;**108**:257601.
241. Burton BP, Nishimatsu T. First principles phase diagram calculations for the system NaNbO<sub>3</sub>–KNbO<sub>3</sub>: can spinodal decomposition generate relaxor ferroelectricity? *Appl Phys Lett* 2007;**91**:092907.
242. Sapper E, Novak N, Jo W, Granzow T, Rödel J. Electric-field – temperature phase diagram of the ferroelectric relaxor system  $(1-x)\text{Bi}_{1/2}\text{Na}_{1/2}\text{TiO}_3 - x\text{BaTiO}_3$  doped with manganese. *J Appl Phys* 2014;**115**:194104.
243. Daniels JE, Jo W, Rödel J, Jones JL. Electric-field-induced phase transformation at a lead-free morphotropic phase boundary: case study in a 93% $(\text{Bi}_{0.5}\text{Na}_{0.5})\text{TiO}_3$ -7%BaTiO<sub>3</sub> piezoelectric ceramic. *Appl Phys Lett* 2009;**95**:032904.
244. Hinterstein M, Knapp M, Hölze LM, Jo W, Cervellino A, Ehrenberg H, et al. Field-induced phase transition in Bi<sub>1/2</sub>Na<sub>1/2</sub>TiO<sub>3</sub>-based lead-free piezoelectric ceramics. *J Appl Cryst* 2010;**43**:1314–21.
245. Kling J, Tan X, Jo W, Kleebe HJ, Fuess H, Rödel J. *In situ* transmission electron microscopy of electric field-triggered reversible domain formation in Bi-based lead-free piezoceramics. *J Am Ceram Soc* 2010;**93**:2452–5.
246. Ma C, Guo H, Beckman SP, Tan X. Creation and destruction of morphotropic phase boundaries through electrical poling: a case study of lead-free piezoelectrics. *Phys Rev Lett* 2012;**109**:107602.
247. Tutuncu G, Li B, Bowman K, Jones J. Domain wall motion and electromechanical strain in lead-free piezoelectrics: insight from the model system  $(1-x)\text{Ba}(\text{Zr}_{0.2}\text{Ti}_{0.8})\text{O}_3 - x(\text{Ba}_{0.7}\text{Ca}_{0.3})\text{TiO}_3$  using in situ high-energy X-ray diffraction during application of electric fields. *J Appl Phys* 2014;**115**:144104.



**Jürgen Rödel** is a Professor in the Department for Materials and Geoscience at Technische Universität Darmstadt (Germany). He received a diploma in Materials Science from Universität Erlangen-Nürnberg and a Ph.D. from the University of California at Berkeley. At TU Darmstadt part of his research is focused on processing of lead-free piezoceramics and high-temperature dielectrics. He received the DFG research award for young scientists (Heinz-Maier-Leibnitz-Price) in 1992 and the DFG research award for senior scientists (Gottfried Wilhelm Leibniz-Price) in 2009. He authored/coauthored more than 250 refereed publications and 4 patents.



**Kyle G. Webber** is an Assistant Professor in the Department for Materials and Geoscience at Technische Universität Darmstadt (Germany). He received a B.S. in Marine Systems Engineering from Maine Maritime Academy in 2003 and a M.S. and Ph.D. in Mechanical Engineering from Georgia Institute of Technology in 2005 and 2008. In 2008, he joined the Institute of Materials Science of the Technische Universität Darmstadt, Germany as a postdoctoral researcher, where he worked on the mechanical properties of ferroelectrics. In 2013 he was awarded the Emmy Noether Research Fellowship by the Deutsche Forschungsgemeinschaft. His primary research interests include temperature dependent ferroelasticity, phase transformations, and fracture of single crystal and polycrystalline ferroelectric. He authored/coauthored more than 45 refereed publications.



**Robert Dittmer** studied materials science with a focus on functional ceramics at the Technische Universität Dresden (Germany). In 2008 he received his diploma with a thesis on fine-scaled piezoelectric fibers made from PZT. From 2009 to 2013 he worked on his dissertation on lead-free piezoceramics at the Technische Universität Darmstadt (Germany). He finished his graduate studies in 2013 and was awarded with the prize for young scientists by the German materials society (DGM). To date he has authored and coauthored more than 20 refereed publications in the field of piezoceramics.



**Wook Jo** is an Associate Professor of the School of Materials Science and Engineering at Ulsan National Institute of Science and Technology (UNIST) in Korea. He received his Master and Ph.D. in Materials Science and Engineering from Seoul National University, Korea. He then moved to Germany, joining Professor Rödel's group as a group leader for processing of ferroelectrics, where he had mainly focused on developing lead-free dielectrics/piezoceramics and on understanding relaxor ferroelectric materials. He authored/co-authored more than 100 refereed papers and 6 patents.



**Masahiko Kimura** is a researcher in Materials Development Group of Murata Manufacturing Company Ltd. in Japan. He graduated Waseda University in 1992, and received a Ph.D. from Nara Institute of Science and Technology in 2008. His main research interests are related to piezoelectric ceramics. He won the Richard M. Fulrath Award from the American Ceramic Society in 2008.



**Dragan Damjanovic** received B.Sc. diploma in Physics from the University of Sarajevo in 1980, and Ph.D. in Ceramics Science from the Pennsylvania State University in 1987. He joined the Ceramics Laboratory, Institute of Materials, at the Swiss Federal Institute of Technology in Lausanne (EPFL) in 1991, where he is now a *professeur titulaire*. His interests include properties and applications of dielectric, piezoelectric and ferroelectric crystals, films, and ceramics. He is an IEEE Fellow, was awarded 2009 Ferroelectrics Recognition Award by the IEEE Ultrasonics, Ferroelectrics and Frequency Control (UFFC) Society and was Distinguished Lecturer for the IEEE UFFC Society for 2010/11. He authored and co-authored more than 200 papers.