An assessment of structural enthalpy and crystallization pathways of $Mg_{65}Zn_{30}Ca_5$ bulk metallic glass and amorphous films

Scott Gleason, David Miskovic, Kevin Laws, Michael Ferry

UNSW Australia
School of Material Science and Engineering

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ABSTRACT

The structural nature and thermal stability of amorphous alloys is highly dependent on the method by which they are produced, i.e. their relaxation rate upon cooling. Both bulk samples and metallic glass films of $Mg_{65}Zn_{30}Ca_5$ were produced by copper mold casting and direct current (DC) magnetron sputtering onto aluminium substrates, respectively. Comparisons between structural enthalpy, crystallization pathways, relaxation and crystallization kinetics of the bulk samples and films were examined by elevated temperature XRD and DSC. Compared with equivalent experiments on the bulk alloy, results for the thin films show distinct differences in structural enthalpy and deviations from the expected crystalline phase evolution, displaying minor peak shifts, failure of some phases to evolve, and variations in the evolution rates.

TABLE OF CONTENTS

1 INTRODUCTION

The structural nature and thermal stability of amorphous alloys is highly dependent on the method by which they are produced, i.e. their relaxation rate upon cooling. Both bulk samples and metallic glass films of $Mg_{65}Zn_{30}Ca_5$ were produced by copper mold casting and direct current (DC) magnetron sputtering onto aluminium substrates, respectively. Comparisons between structural enthalpy, crystallization pathways, relaxation and crystallization kinetics of the bulk samples and films were examined by elevated temperature XRD and DSC. Compared with equivalent experiments on the bulk alloy, results for the thin films show distinct differences in structural enthalpy and deviations from the expected crystalline phase evolution, displaying minor peak shifts, failure of some phases to evolve, and variations in the evolution rates.

Key sources [1, 2] [3]

2 METHOD

2.1 Master alloy

The master alloy of $Mg_{65}Zn_{30}Ca_5$ was produced using high-purity elements of Mg (99.85 wt%), Zn (99.995 wt%), and Ca (99.8 wt%). The alloy was prepared by induction melting in boron nitride coated graphite crucibles, purged with Ar (99.997 vol.% purity) five times, and protected with a circulating Ar atmosphere. Alloy homogeneity was ensured through heating and cooling through a cycle 700°C, 385°C, 650°C, 385°C, 650°C to a casting temperature of 500 °C and 450°C for for injection and gravity casting respectively. Bulk amorphous $Mg_{65}Zn_{30}Ca_5$ plate with a thickness of $XX\mu m$ was produced by copper mold injection casting. The 25.4mm diameter targets were prepared from a cylindrical copper mold gravity casting sectioned to thicknesses of 3.25mm.

2.2 DC magnetron sputtering

Films were produced from an in-house DC magnetron sputtering facility with Ar working gas (99.997 vol.% purity). The power was 15W, typical voltage of $290 - 350V$, nominal chamber temperature of 25 °C, pressure 1 bar, Ar flow 3.01 standard cubic centimetres per minute (SCCM).

2.3 DSC characterization

DSC

2.4 XRD characterization

XRD heated in 5 degree increments at 20 k/min. scans of xx over a range of 30 - 60 2 theta allows 20 minute scans.

3 RESULTS

deposition were for 35 minutes, and saw an average temperature rise of $3-4$ °C during the deposition. Nominal film thickness was about $2.5\mu m$ giving a deposition rate of $1.2nm/s$.

Relaxed differential scanning calorimetry (DSC) of bulk material taken at different heating rates was used to establish the fragility of the $Mg_{65}Zn_{30}Ca_5$ system. From the equations ... a fit of $\beta^{-1} = 1.338E-16e^{5274(\frac{1}{T-T_0})}$ with Adj. $R^2 = 0.972$ was established. This give a $D^* = 20.4$ which using $D^* = 590/(m-16)$ Shuai2014 [4, 5] gives a fragility $m = 44.9$.

4 DISCUSSION

The use of a 60K DSC heating rate compared to the more commonly used 20K rate [sources] shifts peaks for the bulk $Mg_{65}Zn_{30}Ca_5$ alloy about 8 - 15 degrees higher. This higher heating rates were used because crystallization events for the films were different to differentiation at the lower heating rate. Films show little shift to high temperature peaks with increases heating rates, but large shifts with relaxation. Bulk show the opposite behaviour, larger peaks shifts with higher heating rates and little shift with relaxation.

5 CONCLUSIONS

6 ACKNOWLEDGEMENTS

Yu Wang for his assistance with XRD experimentation and Rietveld refinement.

7 REFERENCES

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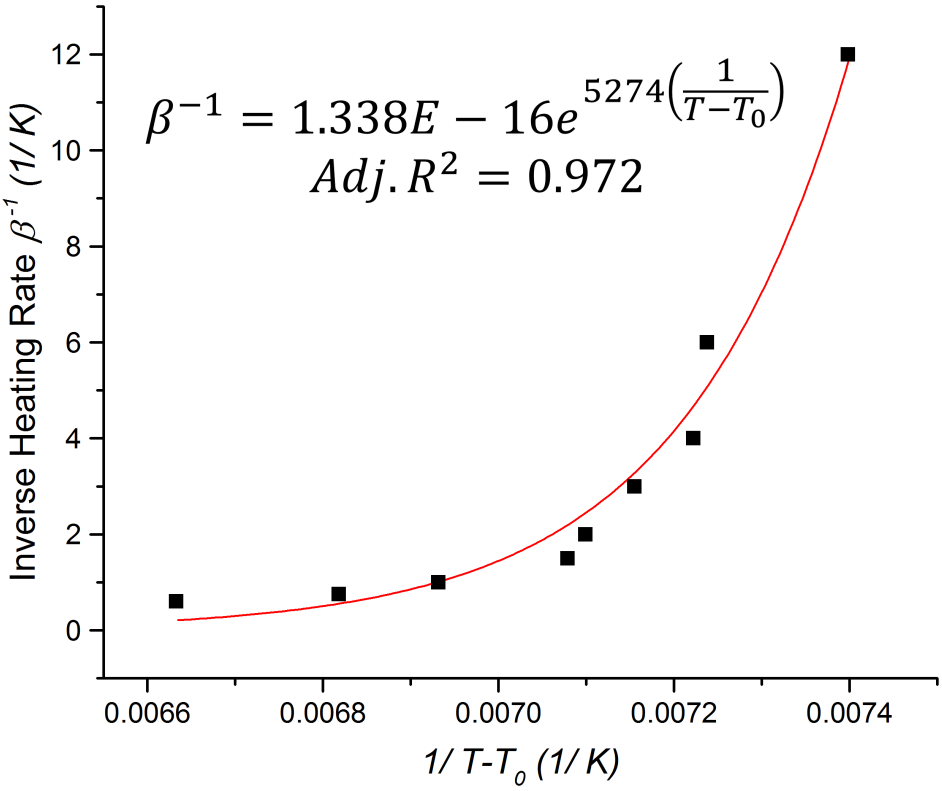
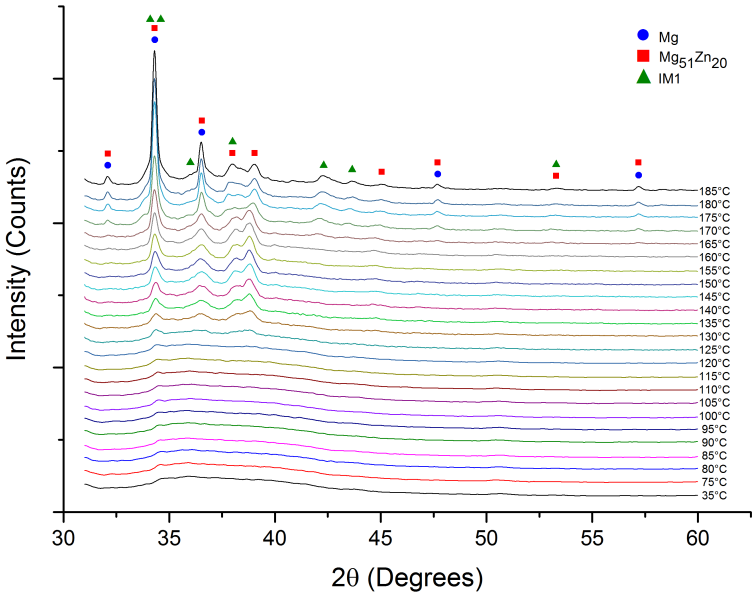
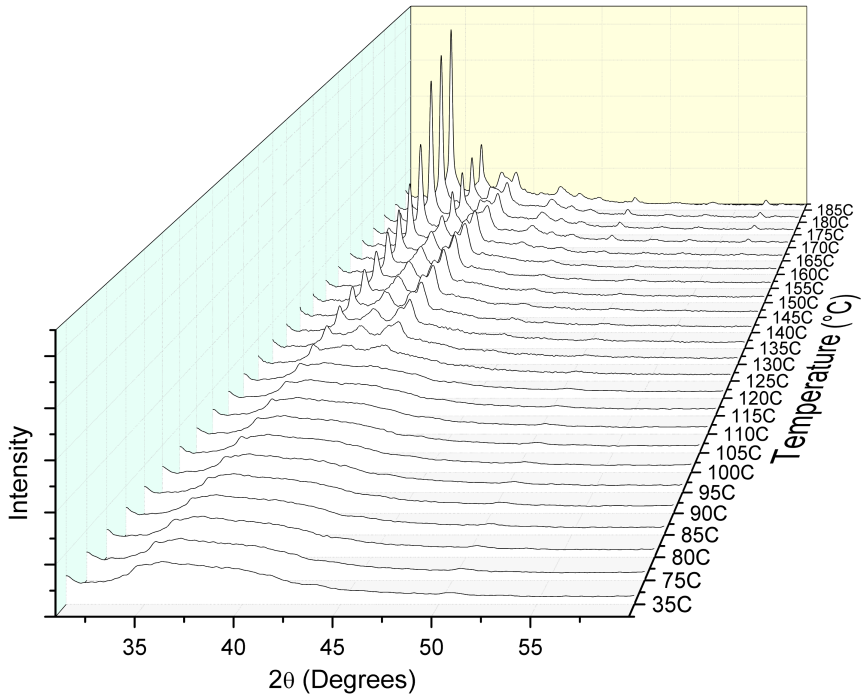


Figure 1: Fitted fragility for the $Mg_{65}Zn_{30}Ca_5$system obtained by DSC at various heating rates

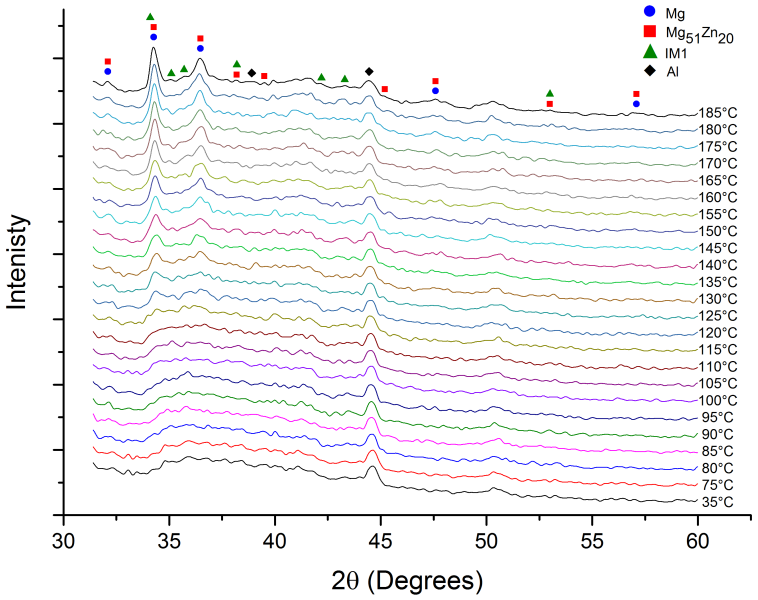


(a)

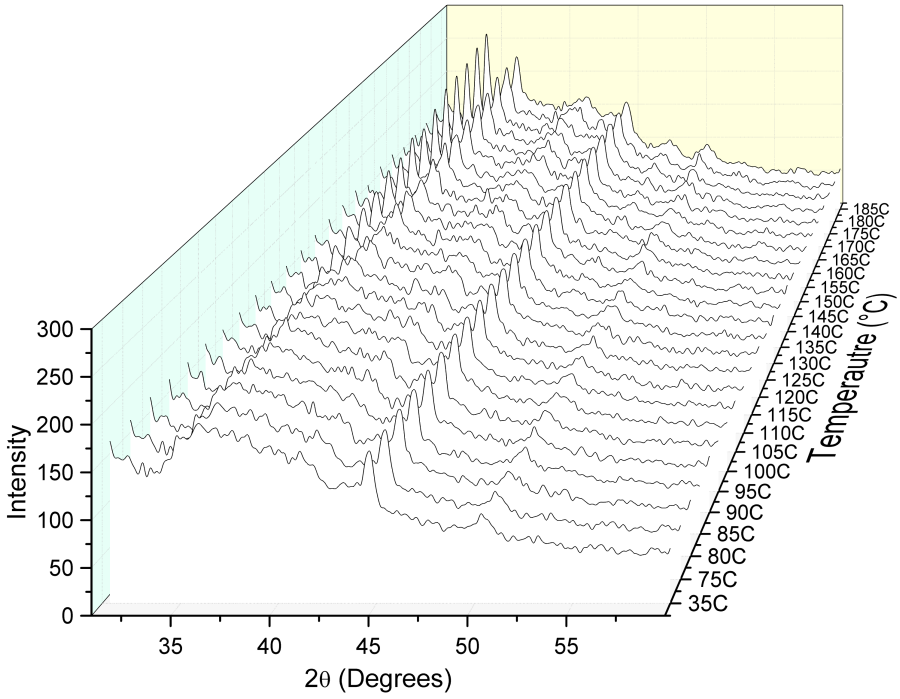


(b)

Figure 2: (a) Stacked X-ray diffraction (XRD) patterns from the incremental heating of bulk $Mg_{65}Zn_{30}Ca_5$. (b) Cascading XRD patterns from the incremental heating of bulk $Mg_{65}Zn_{30}Ca_5$.



(a)



(b)

Figure 3: (a) Stacked XRD patterns from the incremental heating of film $Mg_{65}Zn_{30}Ca_5$. (b) Cascading XRD patterns from the incremental heating of film $Mg_{65}Zn_{30}Ca_5$.

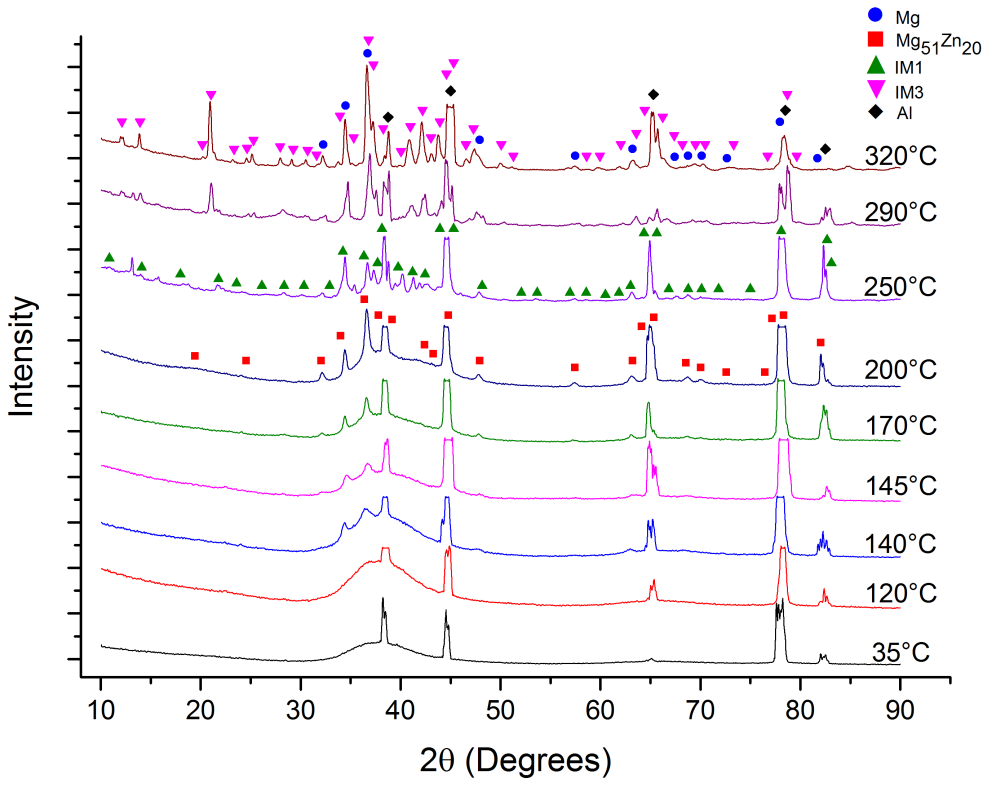


Figure 4: XRD pattern for $Mg_{65}Zn_{30}Ca_5$ heated through several crystallization peaks identified from DSC

DHeating