

# Towards a working tight binding model for Ti – water interface

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## Outline:

1. What is known about Ti interaction with water?
2. From Ti(d) to Ti(sd)
3. Art of the fitting **[Tigany]**
4. Examples of Ti – water simulations
5. Next steps and conclusions

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## How much is known about Ti - water interface?

- Surprisingly, not much. Not much at all.
- We know though that Ti surface oxidises in air (standard conditions) and quickly covers itself with a continuous layer of an unspecified Ti oxide 1–2 nm thick.  
For comparison, Al forms a 2–3 nm layer of alumina (amorphous).  
[many thanks to you, Wikipedia!]
- The oxide layer on Ti can grow up to 25 nm in 4 years time.
- A controllable growth of a thicker TiO<sub>2</sub> layer (up to a few  $\mu\text{m}$ 's) on top of Ti can be done in the procedure called anodising (*e.g.*, wedding rings), but this is a different story.  
Well, perhaps not entirely.
- Experiment (XPS),  $T = 150 - 850 \text{ K}$ :  
O<sub>2</sub> oxidises Ti<sup>0</sup> to Ti<sup>IV</sup>, Ti<sup>III</sup>, and Ti<sup>II</sup>, whereas H<sub>2</sub>O only to Ti<sup>II</sup>.  
Ti<sup>IV</sup>/Ti<sup>III</sup> in case of oxygen, and OH group in case of water are mainly detected in the vicinity of the oxide surface, Ti<sup>II</sup> is found closer to the metal – oxide interface.  
[G. Lu, S.L. Bernasek, J. Schwartz, Surf. Sci. **458**, 80 (2000)]

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## How much is known about Ti - water interface? (cont.)

- For some reason DFT calculations of water – Ti do not appear hugely popular. So far we've managed to find only two papers (!) with overlapping authors (5-layer Ti slab (27 atoms) + 32 H<sub>2</sub>O molecules, VASP) which attempted MD in the presence of electric field using M. Neurock's double reference method [ M. Stancheva *et al.*, J. Electrochem. Soc. **161**, E3188 (2014); I. Betova *et al.*, Comp. Mater. Sci. **171**, 109260 (2020)]
- The electrochemical potential (*electromotive force*) for Ti<sup>++</sup>(aq)|Ti(s) half cell is –1.63 V on the hydrogen scale. For comparison, those for Pt<sup>++</sup>(aq)|Pt(s), Al<sup>+++</sup>(aq)|Al(s), and Mg<sup>++</sup>(aq)|Mg(s) are +1.2 V, –1.66 V, and –2.37 V, respectively [Tony's notes, ver.15, pp. 37–38].
- To summarise:
  - Presumably, as soon as water molecules come in contact with Ti surface, a few of them dissociate and the 'liberated' O atoms are utilised for building some kind of oxide layer.
  - What then happens to H atoms is not entirely clear. They might either (i) dissolve into Ti bulk as H<sup>–</sup>; (ii) take part in the oxide structure; (iii) dissolve into electrolyte as H<sup>+</sup>; or (iv) form neutral H<sub>2</sub> molecules and depart.

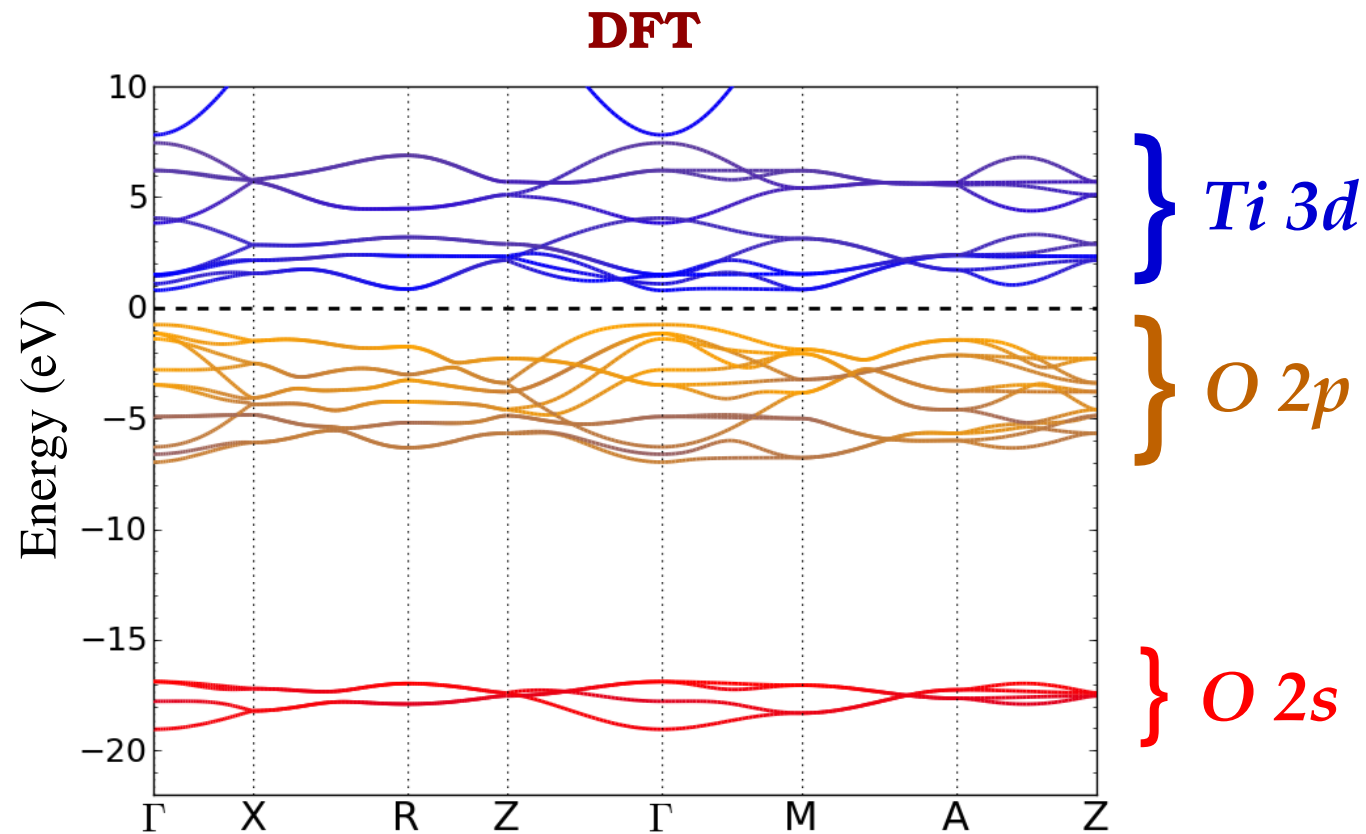
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## Historical overview

The chain of TB models:

1. It all started with Tony's JCP paper on the orthogonal TB model for water [JCP (2011)]
2. The model was further developed and extended to  $\text{TiO}_2$  + water + small organic molecules in 2014 ['three papers', JCP (2014)] which was essentially an effort of the whole Tony's group. Ti in this model had only *d* electrons
3. A couple of years ago Tigany created a new model for pure Ti (bulk phases, phonons, defects, stacking faults, etc.) which was also orthogonal and *d*-only
4. The ambition was to combine models (2) and (3) above so that one could study Ti – water interface.
5. Alas, it didn't work. At least not straight away.

# Titania bands



TiO<sub>2</sub> (rutile) band structure at experimental lattice constants, LDA

# The problem with Ti *d*-only models in a nut shell

The problem appears to originate from having only *d* electrons for Ti

**pure Ti:** DOS of pure Ti is a typical DOS of a transition *sd* metal: a sharp *d*-band on the background of a broad extended *s*-band.  $E_F$  of course falls inside this *d*-band so most properties are defined by *d*-electrons. It is therefore admissible to ignore *s*-states altogether and delegate their contribution to *d*-electrons. Furthermore, according to Tony's experience, *d*-only TB models are often superior to *sd* models.

**d-model:** *s*-orbitals absent;  $N_s(\text{Ti}) = 0$

**titania:**  $\text{TiO}_2$  (rutile or anatase) is an insulator with the valence band made of oxygen *s* and *p* states. States originated from Ti constitute the conduction band  $\simeq 3$  eV above. Ti-*s* states in particular are unoccupied and can be omitted with no impact on the ground state properties of  $\text{TiO}_2$ . Ti *s*-electrons are essential as they are used to fill up the O bands

**d-model:** *s*-orbitals absent;  $N_s(\text{Ti}) = 2$

**intermediate compositions:** Hence, although the neglect of Ti *s*-states seems justified in both limiting cases, pure Ti and  $\text{TiO}_2$ , it's not clear what one should do in intermediate cases where  $0 < \text{O}:\text{Ti} < 2$  and what  $N_s(\text{Ti})$  should be there? Include Ti *s*-orbitals, perhaps?

**sd-model:** *s*-orbitals present;  $N_s(\text{Ti}) = 2$

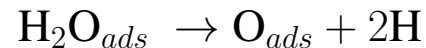
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## Art of fitting: over to Tigany

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## Water dissociation on Ti

- There have been created quite a few *sd* models already, thanks to Tigany. But in all of them water on Ti(0001) dissociates immediately and fully:



- The current model in particular leaves the oxygen atom near the Ti – Ti bridge site and two hydrogen atoms embedded into the Ti surface layer.

Atomic charges are  $Q(\text{O}) = -0.940 |e|$  and  $Q(\text{H}) = -0.989 |e|$ , taken mainly from the top Ti layer in which  $Q(\text{Ti}) = 0.612 \div 0.786 |e|$ .

- **Movie I**

System: Ti(hcp) 2x2 six layer slab (24 at.) with two adsorbed H<sub>2</sub>O molecules, one on each side

Mode: Static relaxation

Symmetry: none

Constraints: Ti atoms are frozen, H and O allowed to relax

- For comparison, in our LDA DFT calculation H<sub>2</sub>O stays non-dissociated during the relaxation (preferred site: Ti – Ti bridge site), but when dissociated ‘by hand’ into either O + 2H (double dissociation) or OH + H (single dissociation), the total energy decreases.



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## Ti - water interface

- Next we move to a larger simulation, our first attempt to emulate the interface between bulk Ti and bulk water. There is no vacuum gap in the cell.
- **Movies II and III**
  - System: 6 layer Ti slab (24 atoms) and 32 water molecules
  - Mode: Static relaxation
  - Symmetry: none
  - Constraints: H and O can relax, substrate Ti atoms are initially frozen, but then released layer-by-layer
- What is the rôle of the dissolved H atoms?
- It appears we can control the extent of the oxidation by just freezing/unfreezing Ti atoms!
- Hopefully, this nice feature does not go away during MD simulations.  
For instance, the next in our plans are NVT runs of a longer cell with all Ti atoms frozen in order to have a good view of the double layer structure.

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## Next steps

- A new version of the model is on its way, thanks to Tigany who works hard on it.
- The most annoying ‘fly’ in the current model is an instability wrt a transition to some ionic structure with lower energy, as is occasionally observed during MD.
- Large scale simulation: MD, double layer
  - A thicker water layer in order to look at the structure of the double layer with all Ti atoms frozen.
  - Also need to establish the cell with the correct density of water.
  - Extraction of Ti into water solution: let us give it a try.
- Electric field created by TB ‘pseudo’ ions, such as Ab atom (an analog of chlorine atom except there is no interaction with other atoms apart from the electrostatic one [Sebastian Budd])
- Hairy probe, of course. Margherita experience will be priceless!

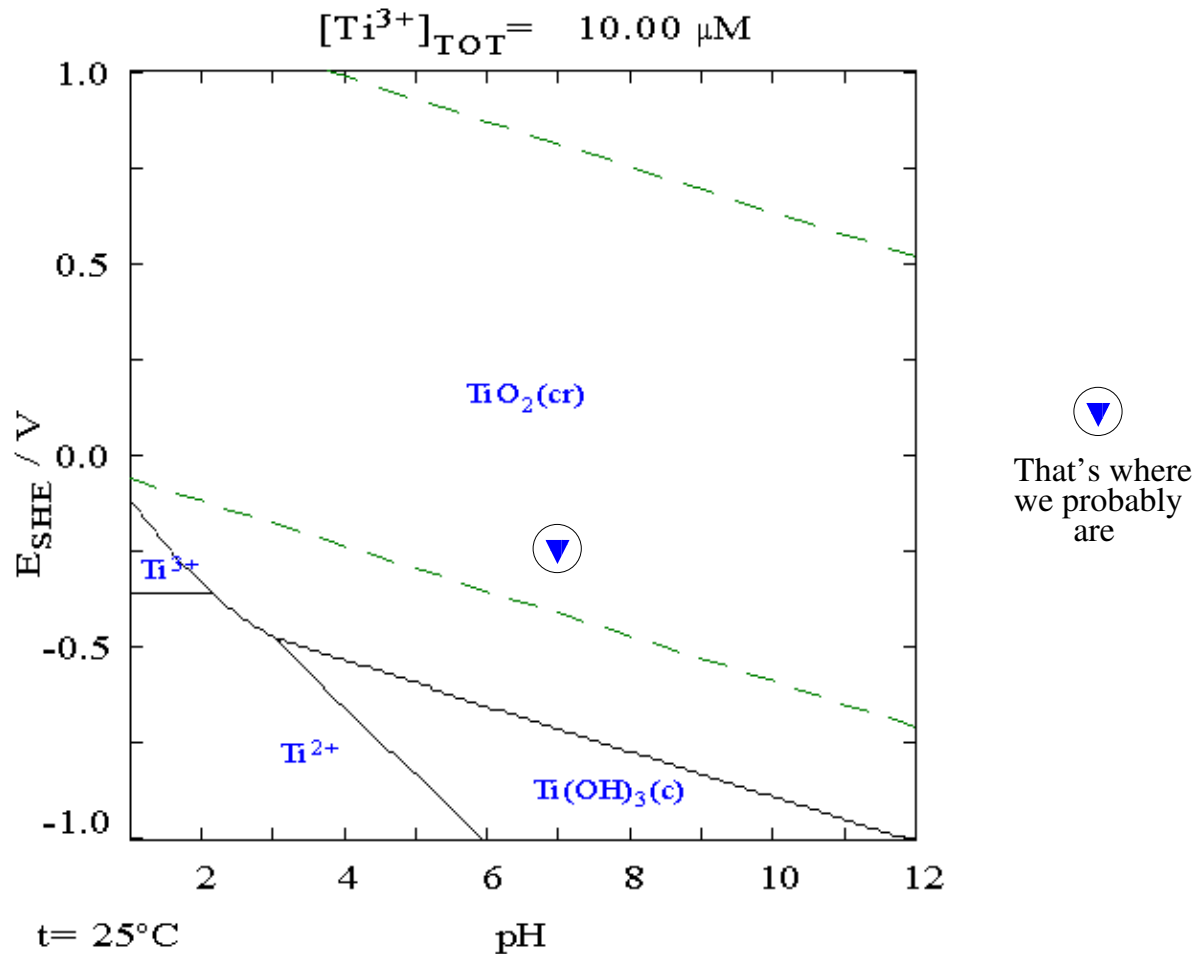
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## Conclusions

1. A reasonably decent (in our opinion) TB model to investigate Ti - water interface has been developed.
2. The breakthrough was due to the switching from Ti(*d*-only) to Ti(*sd*) description.
3. Instability in the bulk Ti(hcp) MD simulations with impurity (O or H) is occasionally observed. Currently we are trying to fix this.
4. The static relaxation of 24 Ti + 32 H<sub>2</sub>O system shows a strong tendency for water molecules near Ti surface to fully dissociate.
5. Released O atoms then try to build a precursor of an oxide layer, whereas the released H atoms either diffuse inside Ti or combine into H<sub>2</sub> molecules and stay near the surface.
6. The growth of the oxide can be controlled in the simulation by freezing Ti atoms.

## Appendix

- Below is the Pourbaix diagram for Ti in water.
- An approximate position of our TB simulations on it is shown with a blue triangle.



Ref: Puigdomenech, Ignasi (2004) Hydra/Medusa Chemical Equilibrium Database and Plotting Software,  
KTH Royal Institute of Technology.