

# Qualitative Evolution of Asymmetric Raman Line-Shape for NanoStructures

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**Abstract** A qualitative evolution of an asymmetric Raman line-shape function from a Lorentzian line-shape is discussed here for application in low dimensional semiconductors. The step-by-step evolution reported here is based on the phonon confinement model which is successfully used in literature to explain the asymmetric Raman line-shape from semiconductor nanostructures. Physical significance of different terms in the theoretical asymmetric Raman line-shape has been explained here. Better understanding of theoretical reasoning behind each term allows one to use the theoretical Raman line-shape without going into the details of theory from first principle. This will enable one to empirically derive a theoretical Raman line-shape function for any material if information about its phonon dispersion relation, size dependence, etc., is known.

**Keywords** Raman line-shape · Phonon scattering · Semiconductors · Nanostructures

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## 1 Introduction

Raman spectroscopy, since its discovery in 1928, is a technique which has been widely used to investigate the vibrational and electronic properties of materials [1, 2]. The crystalline, amorphous or nanocrystalline nature of any material can be ascertained by analysing its Raman scattering data. Raman spectroscopy provides a fast and convenient method to analyze the vibrational properties of crystalline and amorphous semiconductors [3–8]. The physics behind Raman scattering in semiconductors or crystals is based on the inelastic interaction of incidence photons with lattice vibrations or phonons, that are sensitive to internal or external perturbations. In crystalline materials in the bulk form, the Raman scattering is limited to near zone centered phonons due to spectroscopic selection rules. But in the case of nanostructures, this selection rule gets relaxed and phonons other than the zone centered ones also contribute due to confinement of phonons in a crystallite of finite dimension [9–12]. This results in a change in the line-shape of the first order Raman spectrum which is typically Lorentzian for crystalline materials.

Line-shapes of Raman spectra from nano-structures are expected to be intermediate between those of the corresponding crystalline and amorphous materials. For example, a broad band around  $470\text{ cm}^{-1}$  can be observed in the Raman spectra from amorphous Silicon (Si) [13] whereas at the other extreme of perfect crystalline Si, a sharp symmetric Lorentzian peak is observed at  $521\text{ cm}^{-1}$  corresponding to the zone-centered phonon [14, 15]. In the case of Si nanostructures, there is a red-shift in Raman peak position and the line-shape also becomes broad and asymmetric compared to that of a bulk crystalline Si [11, 16–20]. The change in the Raman line-shape parameters (peak position, asymmetry and full width at half maxima or FWHM) is

useful in determining the size, shape and size distribution of nano-structures, since phonon softening and widening of Raman lines are related to the size of nanostructures. Attributing the observed asymmetric Raman line-shape from low dimensional material solely to quantum confinement may be misleading because the line-shape parameters are also dependent on temperature, stress, electron-phonon interaction, etc. A general concept to identify the confinement effect using Raman scattering must be understood. After this generalization, all the factors affecting the Raman line-shape can be taken care of by appropriately modeling the Raman line-shape equation for representation.

Asymmetric Raman line-shape from nanostructured semiconductors are often understood in the framework of the phonon confinement model (PCM) established by Richter et al. [21] and later modified by Campbell et al. [22]. This asymmetric Raman line-shape from nanostructures is mainly observed because the translational symmetry of the crystalline materials is broken at grain boundaries, which results in the appearance of specific surface and interface vibrational contributions [23]. This is more pronounced in reduced dimensional systems. These properties change the frequency of the phonons in the nanostructures, which is reflected in the Raman scattering. According to the PCM, intensity of the first-order Raman scattering can be written as Eq. 1

$$I_0(\omega) = \int_0^1 \frac{\exp(-q^2 L^2 / 4a^2)}{[\omega - \omega(q)]^2 + (\gamma/2)^2} d^n q, \quad (1)$$

where  $q$  is expressed in units of  $2\pi/a$ ,  $a$  being the lattice constant, 0.357 nm. The parameter  $L$  stands for the average size of the nanocrystals.  $\gamma$  being the linewidth of the Si optical phonon in bulk c-Si ( $\sim 4 \text{ cm}^{-1}$ ). ' $n$ ' shows the degree of confinement and may take value of 2 or 3 for 2-dimensional or 3-dimensional confinement, respectively. Following Tubino et al. [24], the dispersion  $\omega(q)$  of the optical phonon in a spherical Si nanocrystal can be taken as

$$\omega^2(q) = A + B \cos\left(\frac{\pi q}{2}\right), \quad (2)$$

where  $A = 1.714 \times 10^5 \text{ cm}^{-2}$  and  $B = 1.000 \times 10^5 \text{ cm}^{-2}$ .

After close analyses of Eq. 1, one can say that it has evolved from a symmetric Lorentzian function with line-shape FWHM of ' $\gamma$ '. However, Eq. 1 has been derived using quantum mechanical concepts, each term in Eq. 1 can be understood in terms of a concept without going into detailed mathematics. This will enable one to incorporate a particular perturbation by appropriately modifying Eq. 1 to get a Raman line-shape for the desired system. In the present study, we have discussed in depth, the evolution of an asymmetric Raman line-shape in nanomaterials starting from a symmetric Raman line-shape from its bulk counterpart. Understanding each term in the asymmetric Raman

line-shape (due to phonon confinement effect) will enable the researchers to explore the possibility of the PCM beyond elemental semiconductor nanomaterials, like Si and Ge, and apply the same to other compound semiconductors like *GaAs*, *TiO<sub>2</sub>*, etc. Although it is applicable to all the materials, in the present study we have done a detailed analysis of first-order Raman line-shape for Si nanostructures as an example.

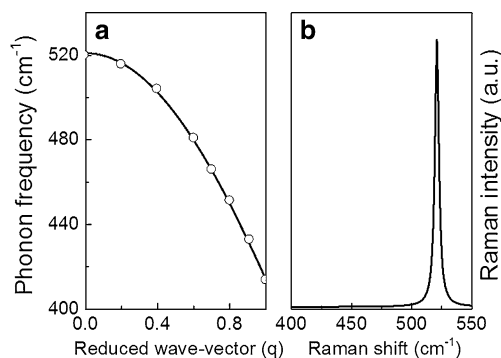
## 2 Analysis and Discussion

As discussed earlier, the Raman line-shape is a manifestation of the Raman active phonon mode of a semiconductor obtained after scattering of light [23, 25]. It is an established fact that a Lorentzian Raman line-shape is observed from a bulk crystalline semiconductor, which is symmetric [23, 26, 27] in nature, whereas that from its nano counterpart is asymmetric [18–20, 28–31]. Figure 1a shows the longitudinal optic (LO) branch of phonon dispersion relation for Si. The discrete points in Fig. 1a shows the data reported in literature for the same [24, 32]. These data points can be represented theoretically as Eq. 2 [33–35].

Figure 1b shows a typical symmetric Raman line-shape from bulk Si represented by a Lorentzian line-shape and can be written as Eq. 3.

$$I_1(\omega) = \frac{1}{[\omega - \omega(0)]^2 + (\gamma/2)^2}, \quad (3)$$

where  $\omega(0)$  is the zone center optical phonon frequency corresponding to  $q = 0$  in Fig. 1a, and ' $\gamma$ ' is the FWHM of the Raman peak. For bulk Si, the values of  $\omega(0) = 521 \text{ cm}^{-1}$  and  $\gamma = 4 \text{ cm}^{-1}$ , are used to plot the Raman line-shape as shown in Fig. 1b. Since only phonons corresponding to  $q = 0$  values participate in Raman scattering, a Lorentzian function with peak position at  $\omega(0)$  is sufficient to represent the Raman line-shape from crystalline materials using Eq. 3.

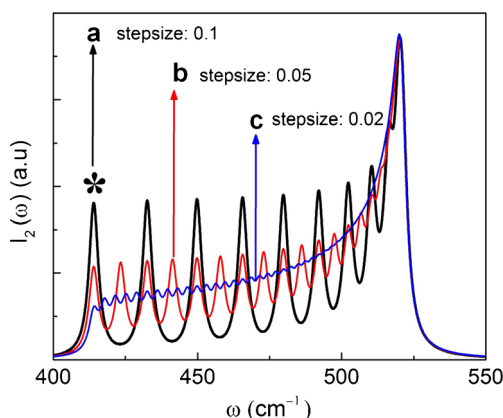


**Fig. 1** **a** Longitudinal optic (LO) branch of phonon dispersion relation for silicon. The discrete points represent the experimental data reported by Tubino et al. [24]; **b** A typical Raman line-shape from bulk Si represented by Eq. 3

The Raman line-shape from nanostructured Si as given in Eq. 1 was derived by Richter et al. [21] and later modified by Campbell et al. [22]. All the terms in Eq. 1 represent a concept and is present with an underlying physics which will be discussed now. At nanoscale, phonons that participate in Raman scattering are confined within the boundary of the nanostructure. Due to the quantum confinement effect, breakdown of Raman selection rule takes place and phonons, away from the zone center ( $q \neq 0$ ), also take part in the Raman scattering [21, 22] in addition to the zone centered ones. As a result, nonzero Raman intensities,  $I_2(\omega)$ , are also expected at lower frequencies (in wave-number units) away from  $\omega(0)$ . To represent this effect in a Raman line-shape function, Lorentzian line-shape corresponding to all frequencies ( $q = 0$  to 1) on the phonon spectrum must be added together. After inclusion of this fact in Eq. 3, the Raman line-shape function can be written qualitatively as Eq. 4 as follows:

$$I_2(\omega) = \sum_q \frac{1}{[\omega - \omega(q)]^2 + (\gamma/2)^2}, \quad (4)$$

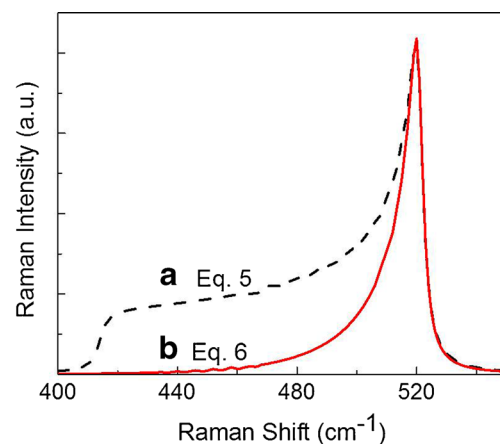
where summation over different values of  $q$  (varying from 0 to 1) represents the equal contributions from all the phonons. Since Eq. 4 involves a summation, variation in  $q$  (0 to 1) can be done in different steps. Raman line-shape in Eq. 4 is plotted in Fig. 2 for three different step sizes *viz.* 0.1, 0.05 and 0.02. It is important to mention here that the plots in Fig. 2 are not actual Raman line-shapes. Spikes in Fig. 2 correspond to the contribution in line-shape from individual phonon corresponding to a particular value of ' $q$ ' and are Lorentzian in nature with peak at a particular value of  $\omega(q)$ . For example, the peak marked with asterisk (\*), centered at  $\sim 414 \text{ cm}^{-1}$ , represents the contribution of phonon represented by  $q = 1$ . For smaller step sizes, the line-shape is continuous instead of spiky as



**Fig. 2** Variation of Raman line-shape intensity as a function of wave-numbers plotted using Eq. 4, by considering contributions from optic phonons at different step sizes on the dispersion curve in Fig. 1a

can be seen in Fig. 2c. The most important feature of the line-shape in Fig. 2c is its asymmetry as compared to the line-shape in Fig. 1b. A careful observation of the evolution from Fig. 2a–c reveals that an asymmetric line-shape arises if all phonons of a branch in phonon dispersion relation (Fig. 1a) take part in the Raman scattering with equal contributions. In addition to this it is also clear from Fig. 2 that spikes are prominent in the lower wavenumber region ( $400 - 475 \text{ cm}^{-1}$ ) as compared to the higher wavenumber region ( $475 - 521 \text{ cm}^{-1}$ ). This observation is directly related to the slope of the phonon dispersion relation in Fig. 1a. The phonon dispersion relation is almost flat near the zone center ( $q = 0$ ), whereas it is monotonically decreasing near the zone edge ( $q = 1$ ), making the dispersion relation nonlinear. Due to this nonlinearity, contributions from the phonons in the vicinity of the zone center add together to result in a higher intensity near  $521 \text{ cm}^{-1}$  (corresponding to  $q = 0$  phonon frequency) forcing the line-shape function to become asymmetric. Line-shapes in Fig. 2 have been plotted by assuming that all the phonons from  $q = 0$  to 1 contribute equally to the Raman intensity and the phonon wave-vector's values are discrete (as represented by steps in the value of  $q$ ). The actual Raman line-shape from nanostructures can now be derived by taking into account the actual contribution from all the phonons and by taking continuous values of phonon wavevectors varying from zone-center to the zone-edge [22].

As discussed above, the actual asymmetric Raman line-shape (as observed from nanostructures) will be derived qualitatively here for the case of Si nanostructures. The contribution of the whole phonon spectrum in the Raman line-shape equation can be incorporated in Eq. 4 by integrating over the full dispersion relation rather than doing the sum-



**Fig. 3** Theoretical Raman line-shape function obtained by considering that all the phonons (from zone-center to zone-edge) take part in Raman scattering. Effect of size is not considered (a) whereas effect of size is considered (b)

mation as done in Eq. 3. As a result, the resultant line-shape equation can be written as Eq. 5 as follows.

$$I_3(\omega) = \int_0^1 \frac{1}{[\omega - \omega(q)]^2 + (\gamma/2)^2} dq \quad (5)$$

Equation 5 is plotted in Fig. 3a, where no spikes are seen in the asymmetric plot due to integration over all the possible values of  $q$ . Almost constant intensity in the  $420 - 480 \text{ cm}^{-1}$  region is due to the hypothesis that all phonons contribute equally or have equal probability to participate in the Raman scattering. Since a nonzero Raman intensity away from  $\omega(0)$  originates due to quantum confinement effect, the actual contribution of a phonon, in Raman scattering, depends on the extent of confinement inside a particular nanocrystal. This may be expressed quantitatively by introducing a weighting function [22] to take care of the actual contribution of a particular phonon. The nature of this weighting function depends on the dimensionality of confinement. For the simple case of spherical nanostructures, an exponential weighting function has been suggested by Campbell et al. [22] for all practical purposes. After introduction of the size dependence, the Raman line-shape is represented by Eq. 6 as follows:

$$I_4(\omega, L) = \int_0^1 \frac{\exp(-q^2 L^2 / 4a^2)}{[\omega - \omega(q)]^2 + (\gamma/2)^2} dq \quad (6)$$

where  $L$  is the size of the nanostructure and  $a$  is the lattice parameter of semiconductor crystal. Equation 6 is plotted by taking the value of  $L = 2.5 \text{ nm}$  and  $a = 0.543 \text{ nm}$  (lattice constant of Si) in Fig. 3b alongside Fig. 3a for comparison. By incorporating the size factor, the line-shape reflects the true asymmetric behavior as observed experimentally in the Raman spectra from nanostructures. Incorporation of the exponential term in Eq. 6 also takes care of the size dependence of the Raman line-shape, which is in consonance with the observed size dependent Raman scattering studies [11, 36–38]. Equation 6 can be understood as the basic Raman line-shape for nanostructured semiconductors in particular and nanostructured materials in general.

Modifications can be done in Eq. 6 to obtain various special Raman line-shapes reported to investigate the effects of stress, temperature, electron-phonon (Fano) effect, quantum confinement effect, etc. Details about these functions are available in the literature [33, 36, 39, 40], and is summarized in supplementary material also. Temperature dependence of phonon dispersion relation can directly be used in Eq. 6 if one wants to investigate the temperature dependent quantum confinement effects [41]. In addition, degree of confinement can also be considered by using the appropriate integral (surface or volume integral) and will result in Eq. 1 from where the discussion of this paper started. If, in a sample, all the nanostructures are not of the same dimension, a distribution function can successfully be introduced

not only to take care of the distribution effect but also to know the exact size distribution available in the sample by theoretical fitting of the experimental data with the modified Raman line-shape.

### 3 Summary and Conclusions

An asymmetric Raman line-shape is observed from semiconductor nanostructures in contrast with a symmetric one from its bulk counterpart. This asymmetric line-shape is used to get information about the shape, size and distribution of nanostructures by theoretically fitting with a theoretical Raman line-shape function. An easy qualitative derivation of an asymmetric Raman line-shape from the symmetric Lorentzian function is suggested here. Different factors affecting the Raman line-shape have been incorporated mathematically in the Lorentzian line-shape function to obtain the asymmetric Raman line-shape for nanomaterials. Physical significance of different terms in the qualitatively derived asymmetric Raman line-shape have been explained for better consonance between theoretical Raman line-shape and experimental Raman scattering data. Better understanding of theoretical background makes the use of Raman line-shape function more versatile and can be used for other semiconductors and materials. If information about the phonon dispersion, size dependence, etc., is known, a Raman line-shape can be derived theoretically from the basic Lorentzian line-shape. Effects of other perturbations (like stress, temperature, etc.) can also be easily incorporated to get a new Raman line-shape.

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