

(Fig. 3d–f) the nanocrystal orientations that are consistent with the spectra and polarizations in Fig. 3a–c. Again, good agreement is obtained.

In Fig. 3g we present the experimental statistics for one-, two- and three-line spectra. One is most common, suggesting that only the lowest sublevel is populated. For the two- and three-line spectra, the measured energy splittings are plotted in Fig. 3h, i. Given three sublevels separated by energies Δ_1 and Δ_2 (Fig. 3i, inset), the average splitting $\bar{\Delta}$ is $0.5(\bar{\Delta}_1 + \bar{\Delta}_2)$, with bars denoting averages. However, two-line spectra can involve any two of the three features, leading to the average $\bar{\Delta}_1/3 + \bar{\Delta}_2/3 + (\bar{\Delta}_1 + \bar{\Delta}_2)/3 = 2(\bar{\Delta}_1 + \bar{\Delta}_2)/3$. We therefore predict a ratio of 1.33 for average measured splittings in two- versus three-line spectra. The experimentally determined ratio of 1.42 ± 0.12 again supports our model.

While we have used cryogenic temperatures to confirm the existence of the bright triplet exciton, its influence on emission remains important at room temperature. Although the splittings are small compared to the thermal energy, the three triplet states (from four sublevels in total) are dipole-allowed and thermally populated, unlike in other nanocrystals^{13–16}. For example, in CdSe nanocrystals only three of eight band-edge sublevels are bright, and these can be poorly populated even at room temperature. This and other effects (Supplementary Information section 5) explain why room-temperature emission from CsPbX₃ perovskite nanocrystals is 20 times faster than in other systems. The emission should be even faster for nanowires and nanoplatelets. Such shapes can further decrease the radiative lifetime, owing to diminished dielectric screening and smaller one- or two-dimensional excitons³⁰.

Although CsPbX₃ nanocrystals are oxidatively stable, their long-term stability may be limited in warm, bright and moist environments without encapsulation to provide thermal and environmental stability. Moreover, the discovery that their lowest exciton is bright reveals criteria for obtaining this phenomenon in other materials. Potential semiconductors should lack inversion symmetry, and one band edge should have *s* symmetry and the other *p*, with the latter affected by strong spin–orbit coupling such that $J_{\text{e,h}} = 1/2$. Finally, the Rashba coefficient for the electron and the hole must be non-zero with the same sign.

Online Content Methods, along with any additional Extended Data display items and Source Data, are available in the online version of the paper; references unique to these sections appear only in the online paper.

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Supplementary Information is available in the online version of the paper.

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Author Contributions This work resulted from ongoing interactions between G.R., M.V.K., D.J.N. and A.L.E. M.A.B., G.R. and T.S. performed the optical experiments, and analysed and interpreted the data with input from D.J.N., R.F.M., P.C.S. and A.L.E. A.L.E. conceived the model and supervised the theoretical research. R.V. calculated the radiative lifetimes and developed the theory for the observed exciton fine structure. R.V. and A.S. developed the four-band model to describe the energy dispersion at the R point and calculated the fine structure using the wavefunction extracted from first-principles calculations. P.C.S. performed the group-theory analysis of the fine structure and selection rules. M.J.M., N.B. and J.L.L. completed the first-principles calculations of the bulk band structures and the band-edge wavefunctions. R.V., P.C.S. and A.L.E. developed the effective exchange and Rashba Hamiltonian that describe the exciton fine structure. J.G.M. and S.G.L. calculated the internal electric fields in spherical and cube-shaped nanocrystals. G.N. prepared the samples and performed electron microscopy under the supervision of M.V.K. A.L.E. and D.J.N. wrote the manuscript with input from all authors.

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