

Investigating Moiré Physics in 2D heterostructures using EMPAD detector:

A study of Charge Density Mapping

Sangers, J. (4645197)

Applied Physics - Master Thesis

TU Delft

24th April 2023

Abstract

Contents

Table of Contents	ii
1 Introduction	1
2 Theory	2
2.1 TMDC/Crystal structure	2
2.1.1 Transition metal dichalcogenides	2
2.1.2 Crystal lattice and band structure	2
2.1.3 Reciprocal lattice and electron diffraction	2
2.1.4 Convergent beam electron diffraction	3
2.2 Moiré	3
3 Methods	4
3.1 Mechanical transfer	4
3.2 TEM / EMPAD / EELS? / HRTEM	4
3.3 The Transmission electron microscope	4
3.3.1 Bright Field	5
3.3.2 Dark Field, Z-contrast and high-resolution TEM	6
3.3.3 Electron microscope pixel-array detector	6
3.3.4 Electron energy-loss spectroscopy	6
3.4 Momentum resolved electron energy-loss spectroscopy	8
3.4.1 Energy filtered transmission electron microscope	8
3.5 Data analysis	8
3.5.1 CoM analysis	8
3.5.2 Charge density analysis	8
3.5.3 Strain analysis	8
4 Results	9
References	10

1 Introduction

1. use of tmdc, QComp, optoelec
2. flat bands

2 Theory

2.1 TMDC/Crystal structure

2.1.1 Transition metal dichalcogenides

2.1.2 Crystal lattice and band structure

An infinitely repeating group of atoms is called an ideal crystal, such a crystal is constructed by attaching the same group of atoms, often called a unit cell, to a lattice. The lattice can be constructed from n independent lattice vectors. $n = 1$ for an atomic chain, $n = 2$ for a two-dimensional monolayer, and, $n = 3$ for a three-dimensional crystal. If no smaller repeating group of atoms can be constructed to fill the lattice then this group of atoms is called the primitive unit cell and the n -independent lattice vectors are then called the primitive translation vectors a_n [1]. Each of the n lattice vectors signifies a direction and length of displacement needed such that the shifted crystal lattice is indistinguishable from the original crystal lattice 1. Lattice vectors are also used to specify the orientation of a crystal plane by denoting where the plane intersects the lattice vector, this procedure allows for unique indexing of crystallographic planes. The use of these planes will be discussed in 2.1.3.

$$\vec{r}' = \vec{r} + u_1\vec{a}_1 + u_2\vec{a}_2 + u_3\vec{a}_3 \quad (1)$$

2.1.3 Reciprocal lattice and electron diffraction

In the previous section the crystal lattice was introduced, and it was mentioned that there were unique planes characterized by the points where they intersect the lattice vectors. In reciprocal space every lattice point is equivalent to one set of these planes. To best understand a crystal, it is helpful to conceptualize it as having two lattices. The first lattice pertains to the organization of the atoms within the crystal's unit cells. The second lattice is a pattern of points that is specific to each crystal and does not correspond to the atom arrangement. Rather, each point in the lattice is linked to a particular set of planes within the crystal [2]. Both lattice constructions are equally valid but are helpful under different circumstances; the reciprocal lattice, for instance, is a useful geometrical construct when talking about diffraction.

The reciprocal lattice, just like the crystal lattice, is constructed by vectors; in the case for the reciprocal lattice these are the reciprocal lattice vectors \vec{b}_n . The reciprocal lattice vectors are constructed from the real-space lattice vectors using equation (2) and satisfy relation (3) with their real-space counterpart. Using these definitions the reciprocal lattice vectors are unique. Any reciprocal vector can now be composed uniquely by a linear combination of the reciprocal lattice vectors, such that any vector is scaled and summed. If the scalars are integers they are the miller indices and correspond to a crystallographic plane.

Scattering off of these planes shows as a series of high-intensity spots in a diffraction pattern image (Figure 1), such an image can be taken in the diffraction mode of a scanning transmission electron microscope (STEM).

$$\vec{b}_i = 2\pi\vec{a}_j \times \vec{a}_k \cdot [\vec{a}_i \cdot (\vec{a}_j \times \vec{a}_k)]^{-1} \quad (2) \quad \vec{b}_i \cdot \vec{a}_j = 2\pi\delta_{ij} \quad (3)$$

In a transmission electron microscope the electrons emanating from the field emission gun are modelled as plane waves. When incident on an atomically thin crystalline sample, the plane waves scatter predictably following the physical criteria: incoming and outgoing electrons beams are plane

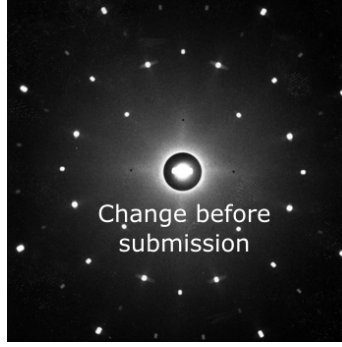


Figure 1: Diffraction Pattern

waves with wave vectors \vec{k}_I and \vec{k}_O for incoming and outgoing waves. The resulting change in wave vector due to the scattering of the sample is then equal to $\vec{K} = \vec{k}_I - \vec{k}_O$. As seen in Figure 2, the outgoing electron beam wavefront is deflected by an angle θ from the incident electron beam such that both are in phase, this angle is the Bragg angle [2]; and using that $|\vec{k}_I| = |\vec{k}_O| = |\vec{K}| = \lambda_e^{-1}$, with λ_e the electron wavelength, the scattering angle can be expressed as:

$$\sin \theta = \frac{|\vec{K}|/2}{|\vec{k}_I|} \quad (4)$$

If both outgoing rays from the same incoming beam wavefront are then in phase, meaning that the extra distance travelled by one of the rays is a multiple of the wavelength, then the following condition holds for the Bragg angle:

$$n\lambda_e = 2d \sin \theta_B \quad (5)$$

This shows that scattering allows for a finite quantized momentum transfer from the electron to the crystal lattice or vice versa. In a crystalline sample this results in bright spots in the diffraction image, where each bright spot can be indexed and attributed to a family of planes in the crystal that facilitate the momentum transfer for the electrons to reach that spot on the detector or phosphor film.

2.1.4 Convergent beam electron diffraction

1. crystal lattice
2. diffraction pattern / reciprocal space

2.2 Moiré

1. moire pattern -> stacking
2. lattice relaxation
3. mini brillouin zone / aligned or antialigned

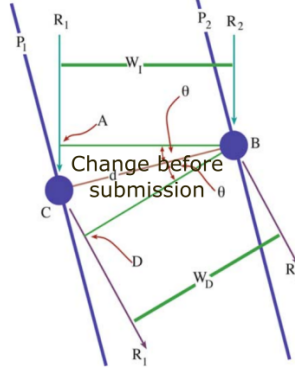


Figure 2: Scattering diagram

4. hybridisation, inter/intra transistions and excitons
5. band bending types, umklapp, flat bands

3 Methods

3.1 Mechanical transfer

Mechanical transfer relates to the method of moving a sample from one substrate to another whilst minimising the amount of imperfections introduced. This transfer is usually from the growth substrate to a target substrate used for an electrical device or to a TEM grid for characterization in a transmission electron microscope.

3.2 TEM / EMPAD / EELS? / HRTEM

1. Electron microscope workings and explanation of all detectors
2. empad detector working / uses
3. CoM for electric and magnetic fields
4. charge density mapping
5. Strain mapping

3.3 The Transmission electron microscope

The Transmission electron microscope (TEM) is a microscope that far exceeds the capabilities of a normal light microscope. Both types of microscope use a series of lenses to magnify the image of a specimen. A normal light microscope can amplify an image up to about $1500\times$ and is limited by the diffraction limit of light. Assuming an average wavelength of 550 nm for green light, a high-end microscope is limited to resolving features 100 nm apart. This limit is insufficient for looking at

atomic structures [3].

An electron microscope circumvents this limit by using electrons, not light, to probe the specimen. Electrons when accelerated have a smaller wavelength than light thus allowing for images with resolved features as small as 0.05 nm. [4] The TEM works by releasing electrons from an electron source and accelerating them to an energy typically expressed in kilo-electronvolt; as shown in equation 6, the higher the accelerating voltage of the microscope the smaller the de Broglie wavelength of an electron, which results in a higher resolving power. Modern electron microscopes accelerate electrons up to 300 keV

$$\lambda_e = h \cdot [2 \cdot e \cdot m_e \cdot V_a]^{-1/2} \quad (6)$$

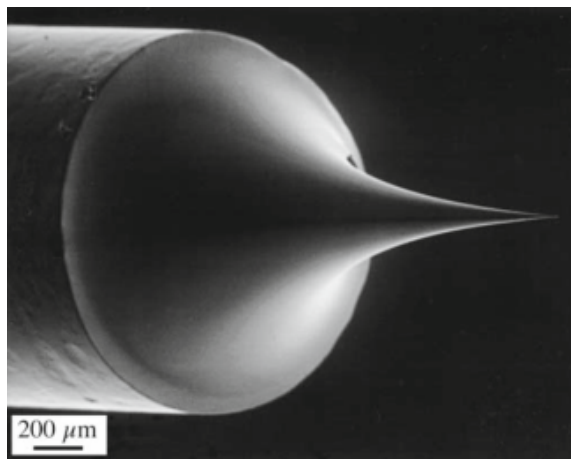


Figure 3: A field emission gun, [2]

In a TEM the electrons are released from a field emission gun (FEG, pictured in Figure 3). The FEG is placed in proximity to two anodes, the first anode is positively charged to several kilovolts such that it extracts electrons from the tip of the FEG, the second anode is charged to the wanted acceleration voltage [2]. FEGs are about three orders of magnitude brighter than thermionic emission electron sources [5]. After being emitted the electrons pass through a monochromator that reduces the energy spread of the emitted electrons. An electron then continues along the optical axis into an illuminating system consisting of multiple electromagnetic lenses, which lenses are activated and to what extent depends on the operating mode of the electron microscope.

3.3.1 Bright Field

A bright field (BF) image of the sample is acquired by using a parallel electron beam. This beam is formed using the lens configuration shown in figure 4. This parallel beam is typically several micrometers in size at magnifications up to 20k-100k \times . In normal operating mode a BF image is captured by a camera looking at a phosphorous screen or by a camera directly, this method of imaging is most analogous to a normal light microscope. The electron beam is focused in such a way that it illuminates the sample with a parallel beam, such a beam is also used for creating the clearest diffraction patterns.

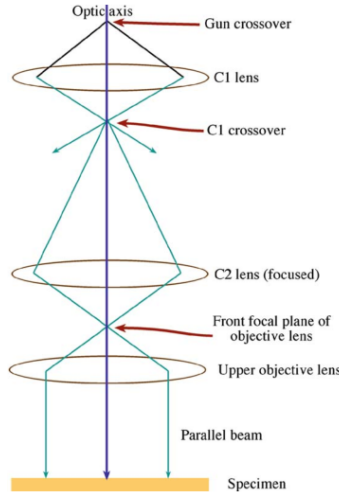


Figure 4: Illumination system for parallel beam.

3.3.2 Dark Field, Z-contrast and high-resolution TEM

For dark field and Z-contrast the electron beam is focused to a small area, this creates a higher intensity electron beam with a probe-like point as can be seen in Figure 5. Since all the electrons are focused on a small area there is no contrast information that can be used to form an image. To form an image using such a beam the probe point needs to be scanned over the sample leading to the term Scanning Transmission Electron Microscopy. Dark field images use electrons scattered away from the optical axis to form an image, to achieve this most STEMs have a series of annular dark field detectors. These ring shaped detectors encircle the central bright field detector and can collect electrons that have been scattered at various angles. Normal dark field detectors collect electrons scattered up to an angle of 50 mrad. The outermost detector is the high-angle annular dark field detector (HAADF) which collects electrons scattered beyond 50 mrad and can be used to create Z-contrast (atomic number Z) or mass-thickness images. The HAADF detector is used since the electrons it collect are almost exclusively incoherently elastically scattered which is proportional to the atomic number Z .

3.3.3 Electron microscope pixel-array detector

3.3.4 Electron energy-loss spectroscopy

In a TEM setup electrons are essentially shot through a sample in which the electrons can either simply pass through or scatter, in the latter scenario there are two possibilities, electrons can scatter elastically or inelastically. Scattering is a result of the interaction between the sampling electrons from the TEM source and the charged particles in the specimen.

When scattering elastically the electrons interact with a nucleus of the specimen whose mass is many times greater than that of the sampling electron, resulting in a small and usually unmeasurable energy transfer. In a crystalline specimen electrons can only be scattered at certain angles due to the crystal structure creating a diffraction pattern of bright spots. In cases of large scattering

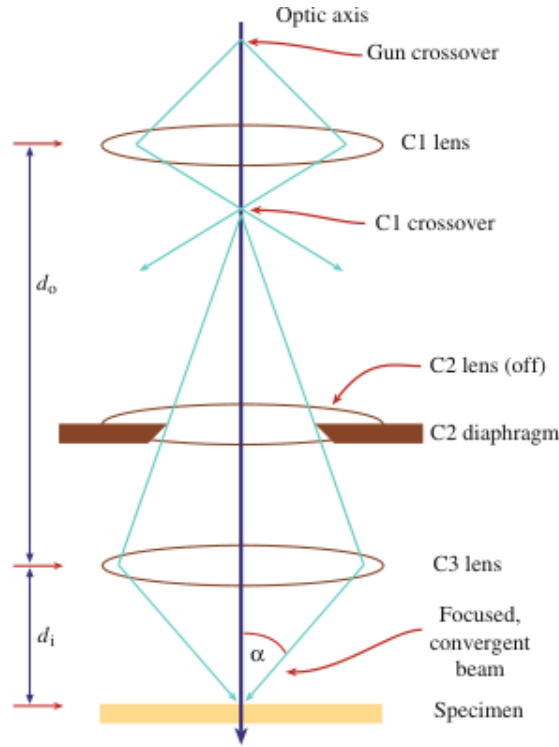


Figure 5: Illumination system for convergent beam.

angles the electron does transfer a significant amount of energy and can even reverse direction, this energy transfer can permanently displace atoms in the crystal structure causing a defect.[6] When the sampling electron interacts with an electron in the specimen's crystal lattice inelastic occurs due to the similarity in mass between the two electrons. The energy transfer of this interaction ranges from a few electronvolts up to multiple hundreds of electronvolts. Inelastic scattering not only results in an energy transfer but also in a momentum transfer as shown in figure ??, the k' -vector shows a scattered electron that deviates from the not scattered electron vector k_0 . The total momentum transfer is the sum of the perpendicular momentum transfer q_{\perp} proportional to the scattering angle θ and the momentum transfer parallel to the undisturbed path due to an energy transfer from the sampling electron to the sample. This parallel momentum transfer is thus proportional to the energy loss of the electron. Figure ?? shows the band structure of the crystalline sample of which the electrons scatter. In this figure two dispersion bands are shown, both bands can be occupied by electrons of certain energies, to excite an electron from the blue band to the red band an electron needs either energy (path t_1) or energy and momentum (path t_2). The needed energy and momentum are transferred from an incident sampling electron in the inelastic interaction. By measuring the energy and momentum of a scattered electron it is possible to piece together all the combinations of energy and momenta transfer possible and thus find the band structure of the sample.

Another form of inelastic scattering is plasmon excitation.

Since the outer-shell electrons of an atom are only weakly bound to the nucleus due to screening effects but are coupled together by electrostatic interaction. These delocalised electrons form an energy band similar to that shown in figure ?? . When a fast-moving sampling electron is shot through the sample all nearby outer-shell electrons are displaced. If the sampling electron's velocity exceeds the fermi speed the displacement of outer-shell electrons creates an oscillating ripple creating waves of alternating positive and negative electric charge, this is known as a plasmon wake.[6]

3.4 Momentum resolved electron energy-loss spectroscopy

Momentum resolved electron energy-loss spectroscopy hereafter abbreviated as MREELS is a TEM imaging technique in which the imaging plane of the CCD camera is placed in the focal point of the imaging lenses and energy filter. This allows the camera to take a diffraction mode image in which the diffraction pattern of the scattering electrons is shown. This is illustrated in figure ?? . In this figure the crystalline sample is shown as the box with the slanted lines that represent the Miller planes of the crystal structure. As shown in the figure all electrons that scatter of the same family of Miller planes get focused on the same region on the imaging sensor, electrons that do not scatter also get focused in one spot at the centre of the diffraction pattern. A diffraction mode image does not image normal space but instead shows reciprocal space which is also the reason this type of image is useful. In a normal image one would attribute lengths to the axes of an image but in a diffraction mode image the separation of features is given by a momentum difference. The separation of the light and dark grey arrows on the imaging plane in figure ?? is thus equal to the difference in perpendicular momentum transfer between scattering electrons (light grey) and electrons that do not scatter (dark grey), a 3D representation is presented in the Appendix. Since the momentum transfer for the electrons that do not scatter is zero the momentum transfer for scattering of a certain family of planes can be determined.

3.4.1 Energy filtered transmission electron microscope

An energy filtered transmission electron microscope is a microscope with an energy filter placed in the optical column of the TEM. Energy filtering is accomplished by the use of electromagnetic prisms such as those shown in figure ?? . These prisms just like ordinary prism disperse the electrons with different wavelengths which are proportional to electron energy. By sliding a slit into the cone of dispersed electrons it is possible to choose a finite range of electron energies to image. The EFTEM setup can be used in conjunction with the MREELS imaging technique to gather information on both the momentum transfer of the electron (via MREELS) and the energy loss associated (via EFTEM) with that momentum transfer.

3.5 Data analysis

3.5.1 CoM analysis

3.5.2 Charge density analysis

3.5.3 Strain analysis

4 Results

References

- [1] C. Kittel, *Introduction to solid state physics*. Brisbane, QLD, Australia: John Wiley and Sons (WIE), 7 ed., Oct. 1995.
- [2] D. B. Williams and C. B. Carter, *Transmission electron microscopy*. New York, NY: Springer, 2 ed., Jan. 2009.
- [3] E. G. van Putten, D. Akbulut, J. Bertolotti, W. L. Vos, A. Lagendijk, and A. P. Mosk, “Scattering lens resolves sub-100 nm structures with visible light,” *Phys. Rev. Lett.*, vol. 106, p. 193905, May 2011.
- [4] C. Kisielowski, B. Freitag, M. Bischoff, H. van Lin, S. Lazar, G. Knippels, P. Tiemeijer, M. van der Stam, S. von Harrach, M. Stekelenburg, and et al., “Detection of single atoms and buried defects in three dimensions by aberration-corrected electron microscope with 0.5-Å information limit,” *Microscopy and Microanalysis*, vol. 14, no. 5, p. 469477, 2008.
- [5]
- [6] R. F. Egerton, “Electron energy-loss spectroscopy in the TEM,” *Reports on Progress in Physics*, vol. 72, p. 016502, dec 2008.