

Computing Class Case Study: Atomistic Modelling of Ionic Conductors

Briefing document

with acknowledgement to

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A1. Overview

This exercise provides an introduction to a class of materials known as fast ion or superionic conductors. Please read through these briefing notes in advance of the class: they will help you understand the scientific background behind it. There will be a separate script to guide you through the details.

The objective is to explore the properties of such materials and characterise their behaviour and in doing so see how Molecular Dynamics can be used to study ionic solids at the atomistic level.

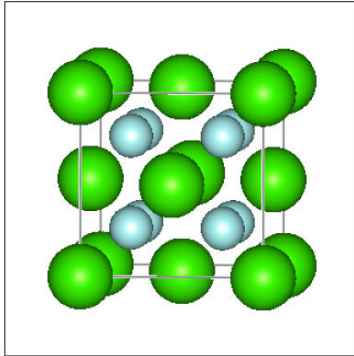
A2. Background

Many inorganic materials conduct an electric current in the solid state. This happens because one or more of the ions is capable of diffusing rapidly through the system while it is still in the solid state.

An example of such a material is Fluorite (CaF_2).

With increasing temperature however the structure shows increasing disorder. It is the nature of this disorder that we are going to investigate and characterise.

A3. Fluorite – (CaF₂)



To build an atomistic model we basically need 3 pieces of information

- 1) The dimensions of the unit cell
- 2) Position and types of ions
- 3) A space group
(description of the symmetry of the ions in the unit cell)

For CaF₂

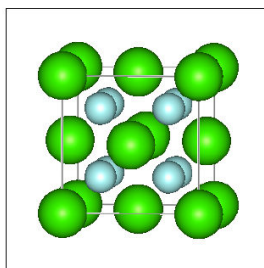
$a=b=c= 5.46\text{\AA}$

Space Group = FM-3M (225)

Ca²⁺ 0.0 0.0 0.0

F⁻ 0.25 0.25 0.25

A4. Born Model of Solids



We use parameterised functions to describe the energy of a system as a function of atoms' positions.

$$V(r_{ij}) = \frac{q_i q_j}{r_{ij}} + \phi(r_{ij})$$

Long range interactions
Coulombic energy

Short range interactions
Van der Waals Bond stretch
Angle Bend Torsion
e.g. Buckingham Potential

$$\phi(r) = A \exp(-r/\rho) - C/r^6$$

Potentials are fitted to experimental properties
or using *ab-initio* calculations

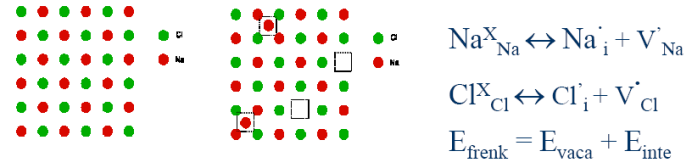
Published well tested models of CaF₂ already exist.

We will be using rigid-ion model *J. Phys: Condens. Matt.* 1:1205-1212 (1989)

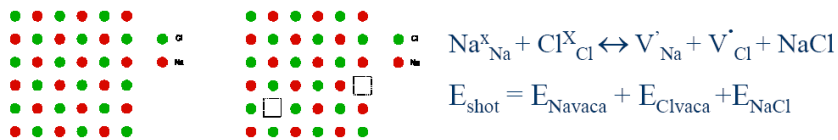
A5.1 Defect Calculations

A quick way of investigating how ions may move within a structure
Model validation -- Can Use Static Methods

Frenkel Defect – Energy required to create an isolated vacancy and interstitial

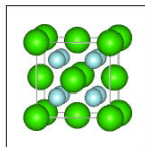


Shottky Defect – Energy Required to remove one formula unit



A5.2 Defect Calculations

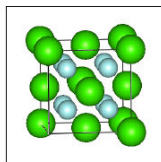
Need to perform 5 energy minimisation calculations



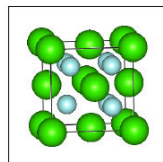
Pure CaF_2 (cell contains 4 CaF_2)

$$E_{\text{latt}} = -107.36\text{eV}$$

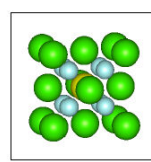
And four defective CaF_2 cells



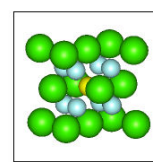
Ca Vaca
-85.92 eV



F Vaca
-102.36eV



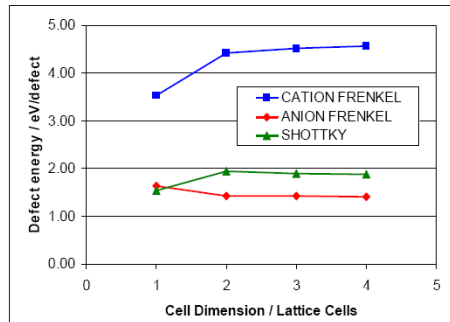
Ca Inte
-121.76eV



F Inte
-109.10eV

A6. Finite Size Effect

The Shottky and Frenkel energies are defined relative to *isolated* defects
In the 1x1x1 calculations the concentration of defects is far from this
High charge relative to size of cell also causes problems
Results suggest Shottky is slightly favoured over Anion Frenkel



Anion Frenkel is now
the most stable intrinsic
defect
Suggest anions will
move through system
via interstitial sites

For anion Frenkel and
Shottky a 2x2x2
supercell is sufficient to
achieve convergence

A7. Molecular Dynamics

Whilst static Energy Minimisation can tell us a lot about our system and the behaviour of defects within them the fact that there is NO temperature included in them and the fact that we have no way of considering the atomic motions directly severely limits what can be done.

Many Properties cannot be explained by a static model
Thermal Properties (Heat Capacity), Thermal Expansion,
Phase Transitions (melting), Transport Properties (Diffusion)

MD provide a simple way of studying such phenomena directly.
This will be the topic of the remainder of the lecture.

A8. Size Effects and Convergence Testing

To ensure periodic images do not interact MD codes require a the short-range cut-off is set to at least half the dimensions of the unit cell

We therefore need a cut-off which ensures our results are accurate but small enough that our system size isn't too large

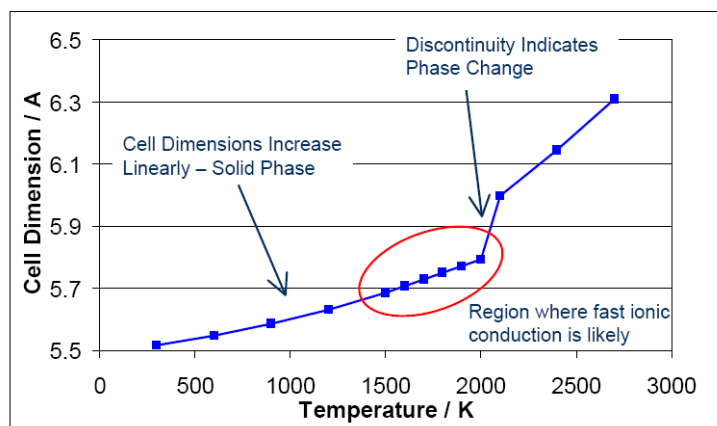
	Energy / KJ/mol	a / Å	volume / Å ³	time / s
1	-10410	5.44	160.89	22
2	-10313	5.52	168.43	306
3	-10312	5.52	168.41	2423
4	-10311	5.52	168.55	11225
5	-10312	5.52	168.37	41144

This system converges rapidly experience has shown that in general as cut-off of 8Å - 10Å is required which therefore requires a cell 16Å - 20Å

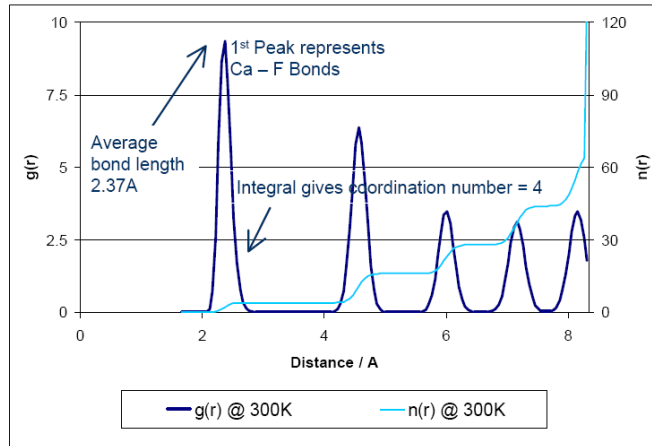
This is only a rule of thumb and convergence test should always be performed when working with a new system

A9. Thermal Expansion

A good test of a model is how the cell dimensions respond to Temperature

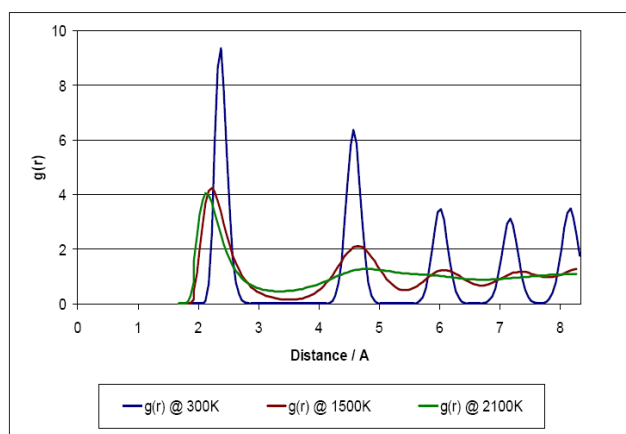


A10.1 Ca---F RDF



Series of sharp peaks characteristic of a crystalline solid

A10.2 Ca---F RDF



At 1500K we still see multiple peaks but they are broader indicating ions are more mobile

Above the melting point we only see the nearest neighbour peak characteristic of a liquid

A11. Characterising Ionic Movement

Mean Square Displacement (MSD)

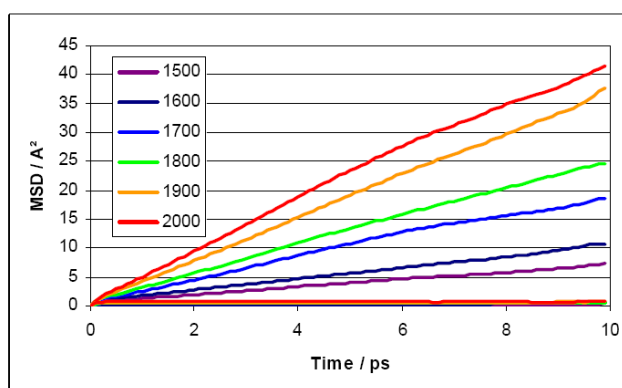
A means of determining the average motion of molecules in a liquid, gas or solid. It is directly related to diffusion, which can be measured experimentally, and is easily calculated in an MD simulation. It is, for this reason, a very important link between modelling and the real world.

Velocity Autocorrelation Function (VACF)

Provides a means of investigating collision processes in molecular systems. Like the mean square displacement, it can also provide information about diffusion. It is capable of distinguishing between solids, liquids and gases from their molecular motion and it reveals the timescales on which molecular collisions occur.

For more information look at DEMOCRITUS on your PC

A12. Mean Square Displacement



Linear plots show F⁻ ions are diffusing
Rate of Diffusion increases with Temperature

Zero Gradient indicates Ca²⁺ cations remain static

A13. Diffusion Coefficient from MSDs

As they travel, the particles are jostled by collisions with other particles which prevent them from following a straight line. If the path is examined in close detail, it will be seen to be a good approximation to a random walk.. It can be shown that the mean square of the distance travelled by particles following a random walk is proportional to the time elapsed.

This relationship can be written as
 $\langle r^2 \rangle = 6 D t + C$

where $\langle r^2 \rangle$ is the mean square distance and t is time. D and C are constants. The constant D is the most important of these and defines the diffusion rate. It is called the diffusion coefficient.

Diffusion in solids usually proceeds by hopping; the atoms jump from place to place rather than simply drift. The energy required for a hop can be calculated from the slope of the Arrhenius plot, the steeper the slope the more energy is required. In liquids, diffusion is relatively easy and so the Arrhenius slope is much less steep.

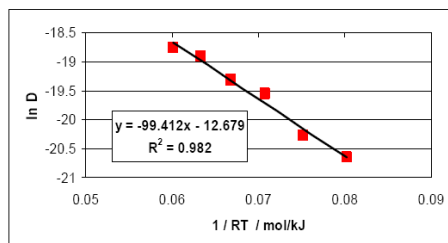
A14. Calculating the Activation Energy: Arrhenius Plot

From the gradient of the MSD plots we can therefore calculate the diffusion coefficient of the F⁻ ions

T / K	1500	1600	1700	1800	1900	2000
D / m ² /s	1.07E-09	1.57E-09	3.19E-09	4.12E-09	6.13E-09	7.12E-09

How D varies with temperature is related to the activation energy:

$$D = A \exp(-E_a/RT) \quad \text{or} \quad \ln D = \ln A - E_a/RT$$



$E_a = 99 \text{ kJ mol}^{-1} \approx 1 \text{ eV}$
 Large energy barrier hence diffusion only occurs at high temperatures

A15. Summary

We have seen how static and dynamic simulation techniques can be used to study solid state systems

We have considered:

What is required to build a model (Crystal structure, potential, etc)

How static calculations can be used to study bulk defects

Finite size effects on static and dynamic systems

The effect of heating on a crystals structure and how this relates to RDF

Ways of studying the Motion of Particles in a system (MSD)

How to calculate diffusion coefficients from MSD and used them to calculate the activation Energy

A16. Tasks

- Devise a plan for remaining days
- Some of the issues to explore
 - Intrinsic defects (noting effect of number and type)
 - Different concentrations of impurities (noting if there is any relationship between properties and their ionic radii and charge)
 - Accuracy of potential model (try different models for Ca-F etc.)
 - External effects (pressure or electric field)
- When considering the effect of defects
 - If negligible effect try increasing the defect concentration dramatically
 - If still no effect could even start from a new structure/composition

A17. Websites

- <http://people.bath.ac.uk/chsscp/teach/>
 - [adv_prac.bho](#)
 - [metadise.bho](#)
- <http://www.ks.uiuc.edu/Research/vmd/>
- <http://accelrys.com/products/discovery-studio/>
- http://www.cse.scitech.ac.uk/ccg/software/DL_POLY/MANUALS/USRMAN2.17.pdf