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Diffusion, growth, and elasticity in batteries A mathematical modelling perspective

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ANNUAL MEETING
INDIAN ACADEMY OF SCIENCES

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Lithium-ion Battery

Background

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Forerunning energy storage medium for portable electronic devices.

- Lithium is the lightest metal known.
- High energy capacity run for long periods between chargings.

Biggest challenge: Use them for electric vehicles? Replace fossil fuels?

- **▶** Even higher energy capacity needed.
- ➤ Search for 'new' battery materials ...

Battery materials

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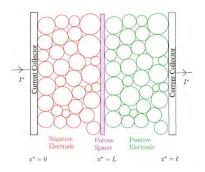
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Most common commercial battery materials:

Negative electrode (Anode): Graphite

Positive electrode (Cathode): Lithium Cobalt Oxide (LiCoO₂)

Electrolyte: Salt of lithium

Figure from G. Richardson, G. Denuault, C. Please, J. Eng. Math 72, 41 (2012).

Silicon for anodes

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Why silicon?

Silicon can accommodate up to 4.4 atoms of Lithium in the fully lithiated state

ightharpoonup Theoretical specific capacity value of 4200 mAhg $^{-1}$ (Graphite: 372 mAhg $^{-1}$)

However ...

Silicon swells up during lithiation - 310% when fully lithiated

⇒ Cyclic charging/discharging leads to mechanical failure (fracture)

A way out ...

Use nanostructured silicon electrode particles

Overview of the problem

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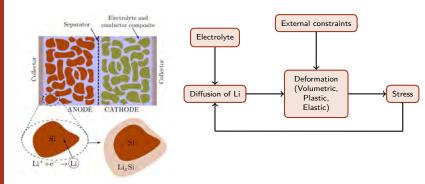
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Particle Level Homogenization Electrode Level

Model Problem

Electrode particle level

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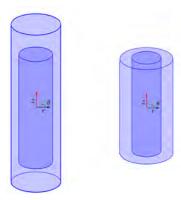


Figure: Left: Unconstrained case. Right: Constrained case – No net axial deformation, edges free to move laterally.

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Diffusion

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pscaling Plastic stretch

Chemical Potential Diffusion

Stress

 $\mathbf{j} = -Dc\mathbf{\nabla}\mu$

Model



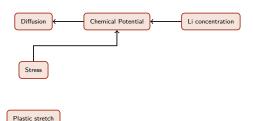
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$$\mathbf{j} = -Dc\mathbf{\nabla}\mu$$

$$\mu = \mu_{\text{conc}} + \mu_{\text{stress}}$$

$$\mu_{\text{conc}} = \mu_{\text{conc}}^{0} + \log(\gamma c)$$

$$\mu_{\text{stress}} = \frac{\partial W}{\partial c}$$

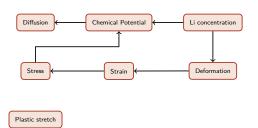
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$$\mathbf{j} = -Dc\mathbf{\nabla}\mu$$

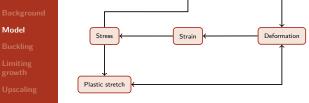
$$\mu = \mu_{\text{conc}} + \mu_{\text{stress}}$$

$$\mu_{\text{conc}} = \mu_{\text{conc}}^{0} + \log(\gamma c)$$

$$\mu_{\text{stress}} = \frac{\partial W}{\partial c}$$

$$\boldsymbol{\sigma}^{0} = \frac{\partial W}{\partial \mathbf{F}} = f(c, \mathbf{E}^{e})$$
$$\mathbf{F} = \mathbf{F}^{p} \mathbf{F}^{e} \mathbf{F}^{c} = \mathbf{I} + \nabla \mathbf{u}$$
$$\mathbf{E}^{e} = \frac{1}{2} \left(\mathbf{F}^{eT} \mathbf{F}^{e} - \mathbf{I} \right)$$
$$\mathbf{F}^{c} = (1 + 3\bar{\eta}c)^{1/3} \mathbf{I}$$

Diffusion



Chemical Potential

Li concentration

$$\mathbf{j} = -Dc\mathbf{\nabla}\mu$$

$$\mu = \mu_{\text{conc}} + \mu_{\text{stress}}$$

$$\mu_{\text{conc}} = \mu_{\text{conc}}^{0} + \log(\gamma c)$$

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$$\mathbf{F}^p = g(\boldsymbol{\sigma}^0)?$$
$$\det(\mathbf{F}^p) = 1$$

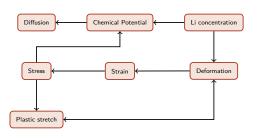
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Net axial force is zero.



Net axial displacement is zero.

$$\mathbf{j} = -Dc\mathbf{\nabla}\mu$$

$$\mu = \mu_{\text{conc}} + \mu_{\text{stress}}$$

$$\mu_{\text{conc}} = \mu_{\text{conc}}^{0} + \log(\gamma c)$$

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$$\mathbf{F}^p = g(\boldsymbol{\sigma^0})?$$
$$\det(\mathbf{F}^p) = 1$$

Governing equations

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Diffusion:

$$\begin{split} \frac{\partial c}{\partial t} &= -\mathbf{Div}\; \boldsymbol{j} \quad \rightarrow \quad \frac{\partial c}{\partial t} = -\frac{\partial j_r}{\partial r} - \frac{j_r}{r}, \\ c(r,0) &= 0, \quad j_r(0,t) = 0, \quad j_r(1,t) = J_0(1-c) \end{split}$$

Mechanical equilibrium:

$$\mathbf{Div} \ \boldsymbol{\sigma}^0 = 0 \quad \rightarrow \quad \frac{\partial \sigma_r^0}{\partial r} + \frac{\sigma_r^0 - \sigma_\theta^0}{r} = 0,$$
$$u(0, t) = 0, \quad \sigma_r^0(1, t) = 0.$$

Plastic stretch evolution:

$$\frac{\partial \lambda_{r,\theta}}{\partial t} = \lambda_{r,\theta} \operatorname{Pf} \left(\frac{\sigma_{\text{eff}}}{\sigma_f} - 1 \right)^m \frac{\tau_{r,\theta}}{||\boldsymbol{\tau}||} H \left(\frac{\sigma_{\text{eff}}}{\sigma_f} - 1 \right),$$
$$\lambda_{r,\theta}(r,0) = 1.$$

Motivation to study buckling

An empirical evidence

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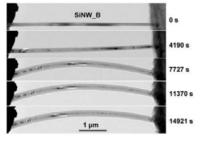


Figure: Buckling of a Si nanowire electrode particle as it undergoes lithiation. (Reproduced from Liu *et al.*, ACS Nano, 7, 1495–1503, 2013.)

Buckling

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General Considerations:

	LOW	HIGH
Length	\checkmark	×
Radius	×	\checkmark
Stiffness	×	\checkmark

For Si electrode particle:



Modifications in the classical buckling criterion

Results

Buckling

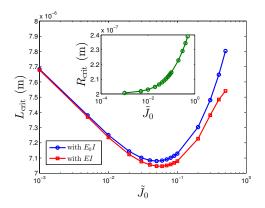
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- Competition between growing stress and growing radius
 - Predominating influence of growing radius

Modified buckling criteria

$$F_{\text{crit}} = \frac{\pi^2 EI}{(KL)^2}$$

E: Modulus of elasticity

$$I=rac{\pi R^4}{4}$$
 : $2^{
m nd}$ moment of area

$$L_{\text{buck}} = \{L : F_{\text{crit}} \text{ exists}\}$$

 $L_{\text{crit}} = \min(L_{\text{buck}})$

Modifications:

$$R = R_0 \left(1 + \frac{\partial u}{\partial r} \right)$$
$$E = E_0 \left(1 + \eta_{Ex_{\text{max}}} c \right)$$

 $E_0 {\it I}: \quad {\rm Only} \,\, R \,\, {\rm changed}$

 $EI: \quad \mathsf{Both}\ E \ \mathsf{and}\ R \ \mathsf{changed}$

An estimate for design

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Approximate the critical compressive force using the yield stress

ightharpoonup Determine critical length

$$\sigma_f \pi R_0^2 = \frac{\pi^2 E_0}{L^2} \left(\frac{\pi R_0^4}{4} \right)$$
 Or,
$$L = \frac{1}{2} \sqrt{\frac{E_0}{\sigma_f}} \pi R_0 \approx 8 \mu \text{m}$$

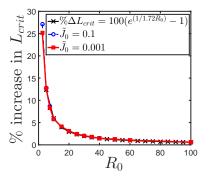
Size-effects

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Young's modulus influenced by lithium concentration
Young's modulus *can also* depend on particle size (radius < 30 nm)

$$E = E_0(1 + \chi_c)(1 + \chi_s)$$

 χ_s determined from Bond-Order-Length-Strength (BOLS) theory



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Sun et al., J. Phys. Condens. Matt. 14, 7781 (2002). Neogi and Chakraborty, J. Appl. Phys. 124, 154302 (2018).

Limiting axial growth

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Is there any \underline{simple} way to limit axial growth? \downarrow Use constraints

Two configurations

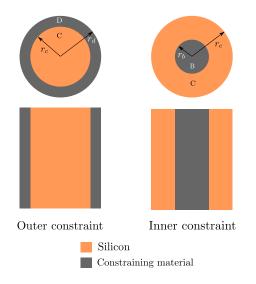
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Phase diagrams

From numerical simulations

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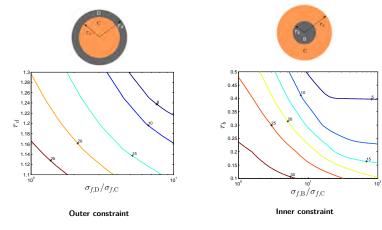
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Yield stress of Region D: $\sigma_{f,D}$

Yield stress of Region C: $\sigma_{f,\,\mathrm{C}}$

Yield stress of Region B: $\sigma_{f,B}$

Contour values indicate percentage increase in length

Ref: J. Chakraborty et al., J. Power Sources 279, 746-758 (2015).

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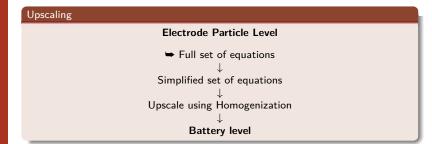
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An open question

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How can we incorporate large strains in the upscaling steps?

→ Apply to silicon anodes

Acknowledgements

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UNIVERSITY OF OXFORD

Prof. S. Jon Chapman Prof. Alain Goriely Prof. Colin P. Please



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THANK YOU