



Short communication

Lithium ion battery electrodes predicted from manufacturing simulations: Assessing the impact of the carbon-binder spatial location on the electrochemical performance

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HIGHLIGHTS

- 4D-resolved lithium ion battery cell model with detailed cathode description.
- Explicit consideration of active material and carbon/binder spatial location.
- Cathode electrode mesostructure calculated by manufacturing process simulation.
- Analysis of the impact of carbon/binder location and properties on cell discharge.
- Application to NMC electrodes and evaluation of operation spatial heterogeneities.

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ABSTRACT

We report a novel computational simulation study of the performance of Lithium Ion Battery (LIB) NMC electrodes for different compositions. The novelty of this work relies on the explicit consideration with three-dimensional resolution of the Active Material (AM) and the Carbon-Binder Domains (CBD) resulting from simulations of the electrode manufacturing processes and considered as spatially separated phases. A multi-phase volumetric mesh is generated and imported into COMSOL Multiphysics using the INNOV meshing algorithm recently published by us. A 4D-resolved electrochemical model is then applied to simulate the electrochemical behavior of the electrode mesostructure upon LIB cell discharge. Several mesostructural parameters are extracted and the differences in the electrochemical response due to the variations of these parameters are investigated. Furthermore, the 4D-resolved electrochemical model allows assessing the impact of CBD spatial location and its transport properties towards Li^+ on the overall electrochemical response, as well as identifying spatial operation heterogeneities inside the electrode. The model assesses then phenomena which are very difficult to investigate based only on experimental approaches, and it aims to become a useful complementary tool of them.

1. Introduction

In the process of changing the paradigm on how human civilization uses and transforms the energy, lithium ion batteries (LIBs) play nowadays a predominant role, in part thanks to several decades of

research and industrial development [1,2]. All the properties of these devices such as performance, cycle life and safety depend on the electrode mesostructure reached at the end of its manufacturing process. Because the electrode mesostructure plays a crucial role in the battery performance [3–5], many experimental studies have been carried out in

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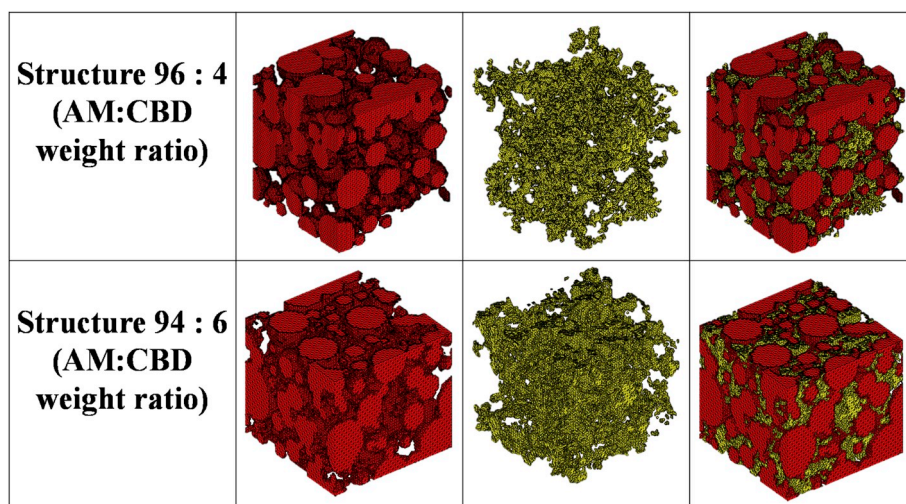
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Scheme 1. INNOV rendering of the meshes of the AM (red) and CBD (yellow) for the compositions (a) 96:4 and (b) 94:6 (weight ratio AM:CBD).

order to unravel the link between the manufacturing parameters and the cell electrochemical response [6–9]. Nonetheless, despite very significant progresses in the development of advanced 3D electrode imaging techniques, there is still a lack of complete understanding of the electrode mesostructural features [4,10]. Their correlation with the electrochemical behavior remains, in part, unclear due to the complexity of the electrode texture and the disparate length scales involved (active material -AM- particles of $\sim 1\text{--}10\text{ }\mu\text{m}$ size and carbon-binder domains -CBD- of $\sim 50\text{--}100\text{ nm}$ size containing carbon additive particles and binder). Due to these challenges, the optimization of the LIB cell performance through its manufacturing process is mainly supported on a time-consuming trial-and-error approach.

Computational modeling gives the promise to overcome this flaw thanks to its predictive capabilities [4]. Based on physics, it has the potential to suggest ways to enhance the performance while reducing the R&D cost and time. Several modeling works have been published in order to study the influence of the electrode mesostructure on the electrochemical performance [11–13]. Such models have been revealed at some extent useful at understanding the discharge/charge behavior of the LIB electrodes. Nevertheless, due to the technical difficulties related to the consideration of the AM and CBD in the electrodes as separated phases, some simplifications are common in the models reported in the literature. These models consider either an electrode mesostructure generated from a stochastic distribution of both phases based on average properties (e.g. weight ratio between AM and CBD), either a hybrid approach with AM reconstructed from imaging techniques and a stochastically-generated CBD [12,14–16]. Such electrode mesostructures are used either to extract effective parameters (such as porosity and tortuosity) injected them in 1D or pseudo-2D models, either in 3D models solving electrochemistry and transport by considering AM and CBD as a single solid phase. The physics related to an explicit representation of CBD, as charge balance and flux between its interfaces with AM and pores is not fully addressed in the literature. Additionally, the impact of CBD properties on the overall performance is not analyzed in depth and no one considers both AM and CBD as explicitly separated phases both originated from the simulation of the electrode manufacturing process. In our previous publications, we have reported a first version of a multiscale computational platform able to predict the influence of some manufacturing parameters on the final electrode performance. Such a platform combines Coarse Grained Molecular Dynamics (CGMD) for the simulation of the slurry, coating and drying, with a 4D-resolved (three spatial dimensions + time) performance model describing electrochemistry and transport mechanisms [5,17]. Despite the implicit consideration of CBD, the 4D-resolved model was able to capture its influence on the discharge performance through effective

parameters (such as the electronic conductivity of an effective solid medium merging CBD with AM) [5,17].

This Short Communication presents a significant advance of our 4D-resolved performance model, where the three-dimensional spatial locations of both AM ($\text{LiNi}_{1/3}\text{Mn}_{1/3}\text{Co}_{1/3}\text{O}_2$ - NMC -here) and CBD within the electrode, as predicted from CGMD simulations, are considered explicitly as separated phases. Such a new feature of the performance model permits the study of the impact of the CBD morphological properties on the overall electrode and cell performance, an aspect which is very difficult to assess based only on electrochemical experiments. The first approach studied here is to consider the CBD as a blocking media towards Li^+ , i.e. to avoid its transport inside CBD. Although it might seem extreme, the primary aim of the CBD is to provide an electronic conductive network, not to transport Li^+ [11,16,18]. However, such an assumption could result in underestimating the performance of the cell by increasing the tortuosity towards Li^+ transport in the electrolyte. The second approach consists in the representation of the CBD as a fully open medium for Li^+ , i.e. Li^+ is able to move in the CBD as freely as in the electrolyte. The Li^+ intercalation is assumed to occur at both AM/electrolyte and AM/CBD interfaces. This second approach might overestimate the performance of the cell by strongly favoring the Li^+ arrival to the AM. The last considered approach is an intermediate case between the first two, where Li^+ can diffuse through the CBD at a lower rate than in the fully open case, mimicking the influence of the CBD microporosity and/or of its swelling driven by the electrolyte. In this intermediate case, the Li^+ intercalation is assumed to occur only at the AM/electrolyte interface. The chosen value for the diffusion coefficient in the partially open case results from an effective consideration of the CBD micro-porosity (details provided in the Supporting Information) and allows a transition from thermodynamic to kinetic driven behavior as the C-rate increases.

2. Methodology

In order to achieve this study, the use of a meshing technique constitutes the first step for the importation of the electrode mesostructures into the 4D-resolved electrochemistry simulator. The meshes have been generated with our *in house* algorithm INNOV, recently published by us [19], which relies on a voxelization technique to create multi-phase volumetric meshes. Meshed electrode mesostructures for compositions 96:4 and 94:6 (AM:CBD) weight ratios, are provided in Scheme 1. It is assumed that all the void space in the mesostructures is filled by the electrolyte. 96:4 and 94:6 compositions have been chosen for this study because they are close to the compositions typically used in the scientific community [18,20]. However, intrinsically, the applicability of the

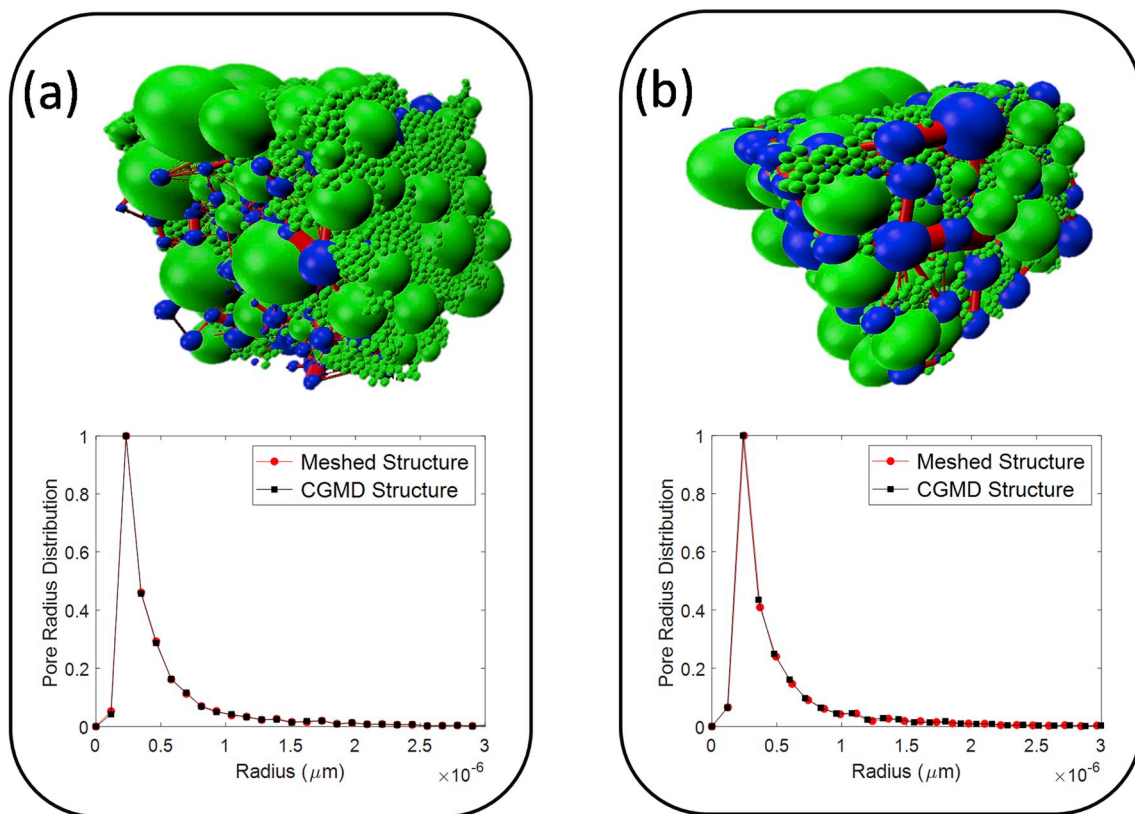


Fig. 1. Pore network of the CGMD structure and pore size distribution of the mesh and CGMD structures for (a) 94:6 and (b) 96:4 electrodes. On the image, the solid particles are represented in green, the throats of the percolation network in red and the cavities in blue. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

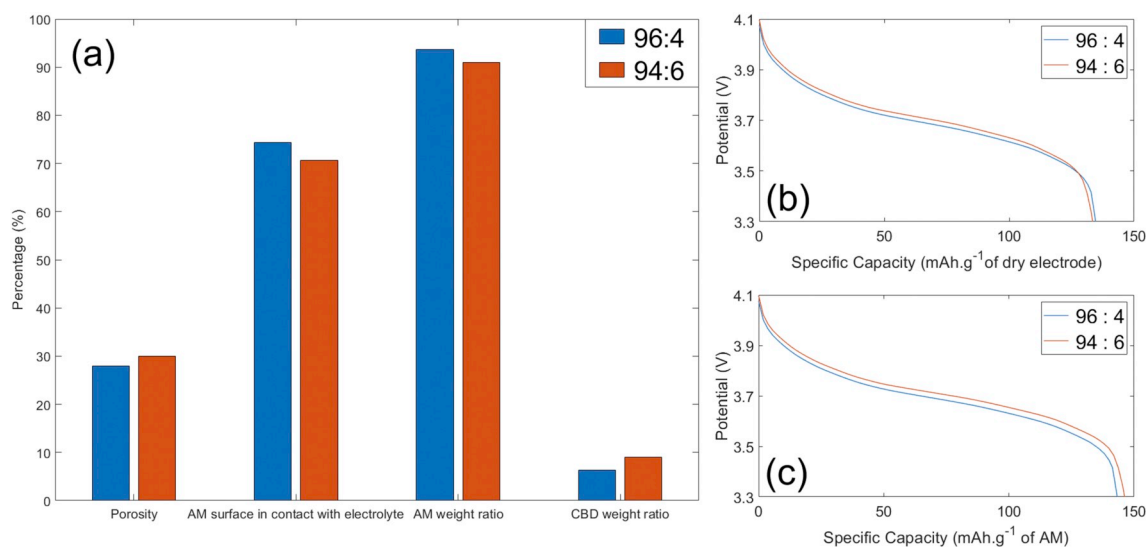


Fig. 2. For both electrodes with 96:4 and 94:6 compositions (a) observables from INNOV, discharge curves at 0.8 C with specific capacity per gram of (b) dry electrode and (c) AM. The simulated discharge curves are extrapolated from 3.4 V to 3.3 V.

meshing algorithm has no limitation in terms of composition. INNOV algorithm is used in this work just as a tool to import the CGMD/tomography structures into our electrochemical simulator: the present communication novelty comes from the numerical simulation of the electrochemical behavior of these electrode mesostructures and the analysis of the CBD impact (with a spatial location explicitly considered for the first time).

The pore networks of the two compositions were extracted through

an algorithm previously reported by us and applied to Li–O₂ batteries [21]. From a binary stack of images, the pore network extraction algorithm divides the pores into throats (cylinders) and cavities (spheres) that fill at best the void volume, while keeping the same porosity. The CGMD mesostructures as well as the associated throats and cavities networks are displayed in Fig. 1. It also provides an insight on the pore radius distribution (PRD), this time for both CGMD and meshed mesostructures. The PRD plots show that the meshing step retains the

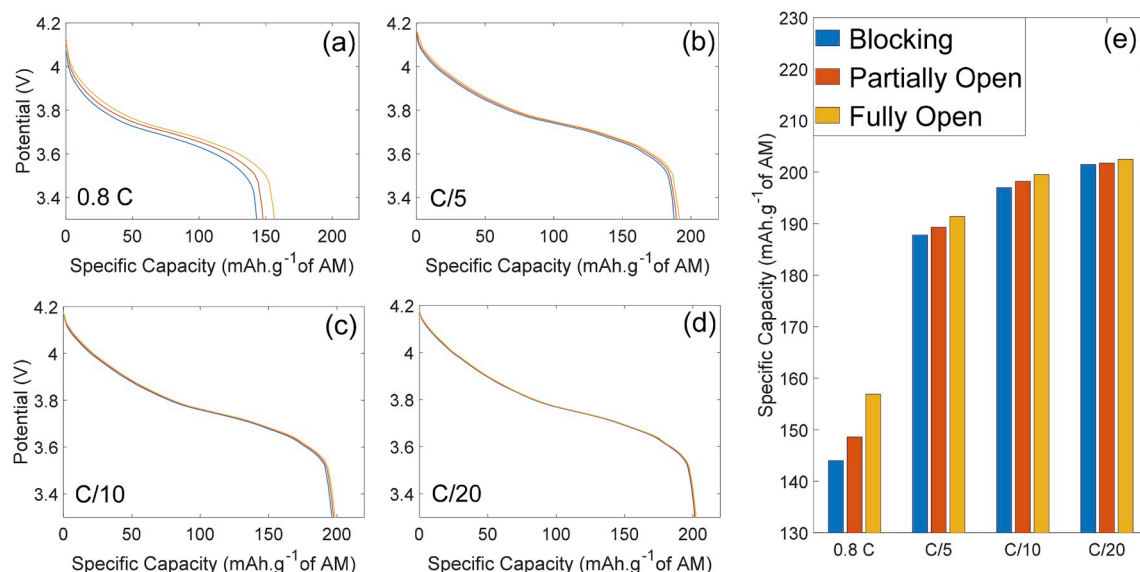


Fig. 3. Discharge curves of the 96:4 electrode along the blocking, partially open and fully open assumptions at (a) 0.8 C, (b) C/5, (c) C/10, (d) C/20 and (e) the histogram of the specific capacities at the end of discharge (3.3 V) for each C-rate. The simulated discharge curves that stopped at 3.4 V are extrapolated in order to reach 3.3 V.

integrity of the pore network since the curve for the CGMD and mesh structures are almost superimposed.

Once the meshing is performed through INNOV, several geometrical features of the electrode can be rationalized (see Fig. 2a). Such features include the porosity, the active surface area (percentage of AM surface area in contact with the electrolyte) and the composition of the electrode to validate the retention of the amount of each phases in the mesh. It is noteworthy that the active surface area strongly depends on the mesh resolution. Indeed, variations in either input image resolution or meshing precision can tremendously change the value of the surface area between the AM and the electrolyte. Consequently, it will also impact the boundary conditions in the 4D-resolved performance model such as the lithium intercalation or electronic flux at the AM/electrolyte and AM/CBD interfaces respectively. Simulation with three different resolutions were carried out in order to study the influence of the mesh size in the electrochemical behavior (see Fig. S5 in the Supporting Information). A change in the resolution of the electrode can derive in an overestimation or underestimation of the AM mass and, in consequence, the specific capacity reached at the end of discharge can be different. Nonetheless, these differences are more significant at higher C-rate where the kinetic factors are dominant. In order to do not have a bias in the final performance, one should use roughly the same length resolution for all the meshes of the mesostructures [5].

We then use our 4D-resolved performance model of a Li-foil/separators/NMC electrode cell to first evaluate how the differences in the NMC electrode mesostructural parameters arise in differences in the electrochemical response. Equations, parameters and other computational details are provided in the Supporting Information.

3. Results and discussions

What it follows from the geometrical features is that a lower content of CBD (4% vs. 6%) leads to a larger active surface area for the AM (74% vs. 70%). The slight differences of these two electrodes is emphasized by the simulated discharge curves at 0.8 C. Here CBD is considered blocking, i.e. Li^+ cannot diffuse through this phase. In fact, in Fig. 2b they show rather similar specific capacities with regard to the dry electrode mass (mass of AM plus CBD). Since the difference between the two compositions is small, a significant gap between capacities is not to be expected. However, Fig. 2c displays a slight difference in specific

capacities in regards to the AM mass. Indeed, the higher AM content electrode has a poorer specific capacity compared to the 94:6 one because its electronic percolation network is less effective than the one of a higher CBD content electrode [22–25]. This evolution of the specific capacity per mass of AM with the CBD content is also reported at the experimental level. [5,23] This demonstrates that the meshing part can shed light on simple trends and that the 4D-resolved performance model is able to capture the expected behavior in regards to the cell performance.

From an experimental point of view, the importance of the carbon additives [26–28] and binder [29] mesostructures in rechargeable batteries has already been acknowledged at the experimental level. The small size of the conductive additive and binder, compared to the active material, provides microporosity where lithium can diffuse/migrate through and that can modify the effective electronic properties of the CBD [11]. Our model is not able yet to reflect explicitly microporosity, so effective transport coefficients are used to capture their impact on Li^+ diffusion/migration through the CBD (see Table S1 in the Supporting Information). On the other hand, changing the Li^+ diffusion coefficient values through CBD can mimic the use of different types of carbon additives and binders. Simulations at different C-rates were carried out for each of the assumptions reported in Fig. 3, namely blocking, partially open and fully open CBD. The C-rates do not exceed 0.8 C because of the current model limitation. The origin comes from the high current density at the current collector arising from low CBD surface area in contact with the solid. Considering that the CGMD simulation used to generate the electrode mesostructures does not take into account explicitly yet the calendaring process, it is believed that the explicit consideration of this step would improve the contact between the current collector and the solid phases. Firstly, the loss of specific capacity while increasing C-rate is captured by the model, i.e. the capacity drops from ca. 201.5 mAh/g at C/20 to 144.1 mAh/g at 0.8 C for the blocking case. Furthermore, the decay of the discharge curves during the voltage drop in Fig. 3a is smoother than the one at lower C-rates which is in agreement with the experimental results [30]. Another trend arising from Fig. 3 is that the specific capacity at a given C-rate is higher for the fully open hypothesis than for the partially open and blocking ones. Such behavior is expected because in the case of the fully open CBD, by assuming a diffusion coefficient higher than it should be, the limitations due to diffusion are underestimated, hence a better performance. On the

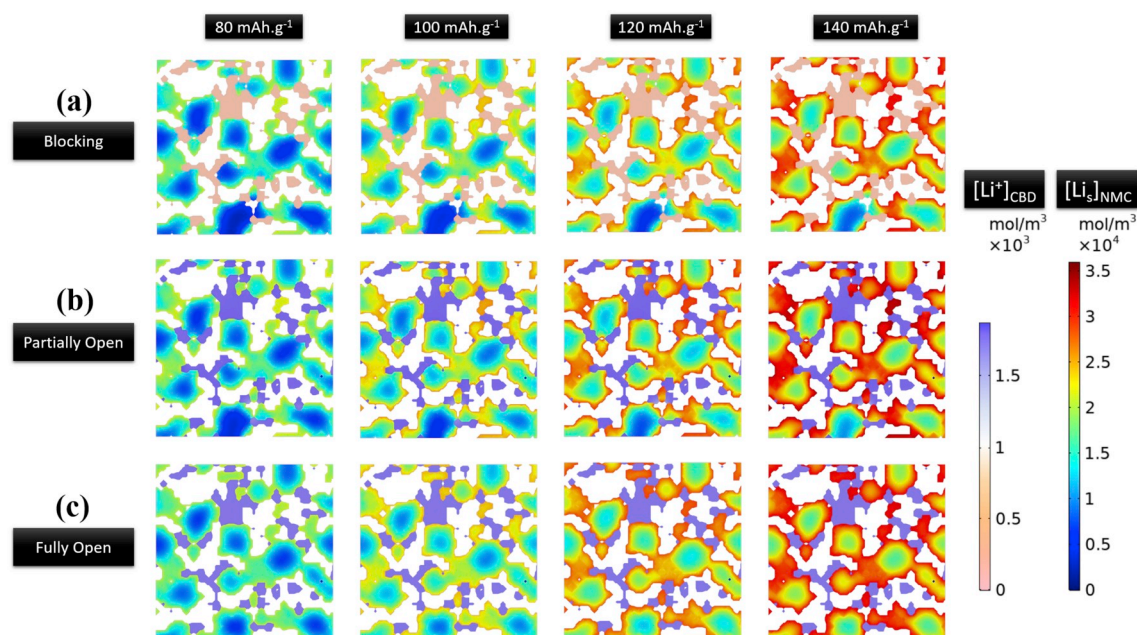


Fig. 4. Snapshots at different depths of discharge of the lithium concentration inside the solid phase for a 96:4 electrode for (a) blocking CBD, (b) partially open CBD and (c) fully open CBD hypothesis at 0.8 C.

contrary, in the case of the blocking CBD, avoiding the access of CBD to Li^+ leads to an increase of the tortuosity of its transport paths, which results in a decrease of the specific capacity, mostly at high C-rate [31, 32]. Obviously, the partially open scenario is in between the previous two assumptions in terms of specific capacity. Interestingly, the difference of specific capacity between the assumptions tends to shrink by decreasing the C-rate. Especially, this is the case for the fully open and partially open cases where the discharges curves are almost similar at low C-rate (Fig. 3d). This emphasizes the shifting from a kinetics-regulated system at high C-rate to a thermodynamics driven one at lower C-rate.

The different approaches regarding CBD can also be characterized by quantifying the lithium (Li^+ and Li(s)) inside the CBD and the AM. Fig. 4 illustrates how each assumption impacts the distribution of lithium in the system at 0.8 C. It is evident that in the case of blocking CBD the concentration inside the CBD will not change during the discharge because Li^+ cannot diffuse through it. Since CBD behaves like an insulating wall towards Li^+ , the latter tends to be accumulated in some regions in the active material leading to steeper concentration gradients, hence a higher polarization resulting in poorer specific capacities. Assuming fully open CBD facilitates the diffusion and the homogenization of Li(s) concentration in the AM. There are smoother concentration gradients thanks to these new diffusion pathways which implies a better cell performance. The partially open approach provides an intermediate scenario where features under both hypotheses can be observed. Indeed, Li^+ concentration in CBD is obviously higher than in the case of blocking CBD however there are also some sharp Li(s) concentration gradients, e. g., in the lower edge of Fig. 4. Therefore, the model is able to capture the expected trends regarding the different hypotheses: it predicts heterogeneous intercalation of lithium in the AM when the CBD is blocking and the homogenization when the CBD is fully open.

4. Conclusions

In this work, the CBD morphology inside the NMC electrode was explicitly considered in a 4D-resolved LIB electrochemical model and its impact on the overall performance was investigated. This implementation of electrode mesostructures calculated by using the CGMD simulation of their manufacturing process was carried out using the meshing

algorithm INNOV. An interesting number of data is obtained through the meshing step such as the pore size distribution, the porosity or the specific active surface area. It has been shown that the electrode mesostructure differences also lead to differences in the electrochemical performances upon discharge. The influence of the spatial location of CBD within the NMC electrode on the LIB cell performance was investigated under three different hypotheses: (i) blocking CBD, (ii) partially open CBD and (iii) fully open CBD. The expected trends were captured by the model, i.e. the first approach tends to decrease the capacity by hindering the diffusion through an increase of the tortuosity of Li^+ transport in the electrolyte. The fully open assumption tends to overestimate the capacity by facilitating Li^+ diffusion whereas the partially open scenario provides an intermediate behavior. This has been further confirmed by the calculated spatial distribution of lithium that sheds light on the heterogeneous intercalation in the AM when CBD is blocking Li^+ transport, creating steep gradients in the regions in contact with the CBD phase, leading to high polarization and poor specific capacity. When the CBD allows the transport of Li^+ under the partially open and fully open hypotheses, the Li(s) concentration in the AM is more homogeneous and a better electrode utilization is achieved. As far as we know, this is the first time that a 4D model predicting explicitly the CBD spatial location impact on the electrochemical response is reported. Further investigations are planned, including performing simulations with electrode mesostructures with other compositions ranges, using higher C-rates or with mesostructures from tomography imaging: for the latter, a proof-of-concept simulation result is presented and discussed in the Supporting Information. This model paves the way to more sophisticated studies, for instance considering the swelling of CBD depending on the electrolyte salt concentration using a dynamic mesh and focusing on the impact of solid/cathode electrolyte interphases on the cell aging.

Declaration of competing interest

The authors declare no competing financial interests.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jpowsour.2019.227285>.

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