Friedrich Schiller Universität Jena PAF

Dissertation

High-Fluence Ion Beam Irradiation of Semiconductor Nanowires

Andreas Johannes

März 2015

Abstract

Hier alles Bla

Contents

1	Intr	oduction	1
2	High Doping Concentrations in Nanowires		2
	2.1	Doping and Sputtering	2
	2.2	nano-XRF on single nanowires	5
	2.3	Pseudo-dynamic simulation	7
	2.4	Discussion of relevant effects	8
3	Sun	nmary and Outlook	12

1 Introduction

2 High Doping Concentrations in Nanowires

This chapter will discuss the concentration of dopants incorporated into ion irradiated nanowires. The simulations and experiments presented in this chapter where all performed with $175 \, keV \, Mn^+$ irradiated ZnO nanowires, however, the effects are easily applied to other material combinations. Some of the first results were published in reference [JNP+14].

2.1 Doping and Sputtering

With iradiana the distribution of the places where the ions come to rest gives the profile of the concentration of dopants per fluence. Locally the concentration $[atoms/cm^3]$ increases a certain amount per fluence $[ions/cm^2]$, leading to the somewhat awkward unit of for the doping efficacy $[(atoms/cm^3)/(ions/cm^2)]$. An example of the dopant distribution simulated with iradina is shown in figure 2.1a for the irradiation of a ZnO nanowire with $175 \, keV \, Mn^+$. The ions enter the y-z plane at random locations and at an angle of 45° to the z-axis, which is periodically continued outside the plane of the image. It is clear that a homogeneous doping profile is not easy to obtain for the irradiation of a nanowire from one side. As with the creation of a box profile in bulk irradiation, multiple irradiation steps with varying energy are required. Note that an ion energy of

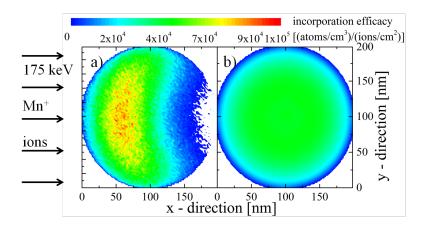


Figure 2.1: a) Color plot of the increase in concentration per fluence for the irradiation of a ZnO nanowire with $175\,keV\,Mn^+$ ions at an angle of 45° to the z-axis. The energy was selected so that the rotation of this profile produces a radially homogeneous dopant distribution, as shown in b). The mean dopant incorporation efficacy is $3.6 \cdot 10^4 \, (atoms/cm^3)/(ion/cm^2)$.

 $175\,keV$ is obviously not enough to permeate the whole nanowire diameter of $200\,nm$, so that an additional irradiation with higher ion energy would be required to obtain homogeneous doping. Rotating the nanowire under the ion beam is a much easier way of increasing homogeneity of the doping profile. Figure 2.1b shows the local dopant incorporation efficacy for the rotation of the profile show in 2.1a. Irradiation with a single, relatively low ion energy produces a homogeneous doping profile.

As lower energy ions have lower ranges, there are fewer paths that cause the ion to leave the nanowire, particularly in the forward direction. Therefore, the first advantage of decreasing the ion energy is that the doping efficacy is larger for lower ion energies, so a lower irradiation fluence is required to achieve doping at a desired level. Furthermore, lower ion energy impacts also produce less damage in the irradiated matrix. Together with an optimal irradiation temperature, the rotated irradia-

2 High Doping Concentrations in Nanowires

tion was utilized to improve the magnetic properties of Mn^+ irradiated GaAs nanowires [BMB+11, PKB+12, Bor12, KPJ+13, PKJ+14].

2.2 nano-XRF on single nanowires

The expected non-linear increase in doping concentration with the ion fluence was first investigated on ZnO nanowire samples grown in Jena, such as the one shown in figure ??a. The nanowires were transferred onto the carbon-foil of a Cu TEM grid by imprinting after the rotated irradiation with 0.24, 0.48, 0.95 and $1.9 \cdot 10^{17} ions/cm^2 Mn^+$ ions at 175 keV; corresponding to Mn/Zn ratios of 0.02, 0.04, 0.08 and 0.16, as extrapolated from the mean doping efficacy obtained from the iradina simulation.

Figure 2.2a shows a SEM image of one of the Mn^+ irradiated ZnOnanowires investigated by nano-XRF at the ESRF. At one point the nanowire shows some damage where the exposure to the XRF-beam was prolonged during the navigation on the sample. Also the track of the intense, focused X-ray beam can be seen on the carbon foil by some redeposition of material. All in all, the damage to the nanowire is, however, not large enough to have an effect on the quantification, especially considering that this particular nanowire was selected as it showed the most pronounced effects. In 2.2b a map of the detected X-ray intensity clearly shows the nanowire. The XRF spectrum collected for one of the scans indicated in the SEM image 2.2a is shown in 2.2c. The number of counts for a single scan is comfortably sufficient to quantify the Mn and Zn content. The average concentration for a nanowire was determined by fitting the sum XRF-spectrum of all scans across the nanowire. The Mn/Zn ratio is plotted over the position along the nanowire for the four nominal concentrations in figure 2.2d. Clearly there is a significant gradient in the Mn concentration along the nanowire length. The maximum Mn/Zn ratio was always found at the tip of the nanowire, which was identifiable in the SEM images by the slight tapering of the nanowires. The Mn/Zn ratio for both the sum of all scans, as well as the scan at the tip showing the maximum Mn/Zn ratio

is plotted in 2.2e alongside the nominal ratio extrapolated from *iradina* simulations.

Two pieces of information can be gained from these results. First, the nanowires on the sample clearly shadowed each other from the ion beam, leading to the pronounced Mn concentration gradient. The shadowing is least at the tips of the nanowires, therefore the corresponding data points are the closest to the simulated situation. This emphasizes the second point, that the increase in Mn concentration with the ion fluence is much stronger than the linear extrapolation from static simulations. The assumption underlying the doping efficacy gained from the earlier simulations and using it to calculate the required fluence for a desired doping concentration is that the concentration increases linearly with the irradiated fluence. However, this is only true in the absence of sputtering. Sputtering erodes the target nanowire at the same time as ions are incorporated. It thus leads to a non-linear increase in the concentration of dopants with the irradiated fluence. To separate these two effects the irradiation and quantification has to be repeated with nanowires with a sparser lateral distribution, as shown in ??b. These were kindly provided by Dr. Helena Franke from the University Leipzig.

The same experimental procedure was followed to investigate the Mn/Zn ratio with the sparser nanowire samples, only rounding the rotated irradiation fluence to 0.25, 0.5, 1 and $2 \cdot 10^{17} ions/cm^2 Mn^+$. The results from the nano-XRF quantification of these nanowires is shown in figure 2.3. The Mn/Zn ratio plotted against the nanowire length in 2.3a no longer shows any gradient. As these wires were individually transferred to the lacy carbon TEM grid, they could be investigated by SEM before and after irradiation. The diameter of the nanowire irradiated with the highest fluence was reduced from 202 nm to 93 nm by sputtering, while the lower fluences produced lower reductions in diameter, as expected. From these diameter reductions the sputter yield can again be calculated, yielding

values in the range of 5 - 20. As seen in the dedicated study on sputtering these values have a very large spread. Also the Mn/Zn ratio for the nanowires irradiated with higher fluences shows a significant spread due to the fact that the thinned nanowires have a much smaller volume and thus give a lower XRF signal. Added to this, the thinner wires could only be attached to the lacy carbon loosely, so that they drifted much more during the XRF scans making it impossible to increase the integration time significantly to compensate for the lower signal. Nevertheless, the average Mn/Zn ratio is accurate to within ± 0.01 , as it is based on the sum of all spectra including a sufficiently large number of counts. The average values for all irradiated fluences is plotted in 2.3b against the irradiated fluence. As with the denser nanowire sample, again the increase in the Mn concentration is much larger and non-linear than the simple linear extrapolation from the *iradina* simulation. Now we can be sure that the incorporation is not effected by the shadowing of the nanowires amongst themselves.

2.3 Pseudo-dynamic simulation

The direct simulation of the effect of sputtering on the incorporation of dopants into nanowires requires a dynamic simulation program, which also considers the three dimensional geometry of the target. As such software is not currently openly available, a step-by-step investigation using results from static simulations will be undertaken to discuss the observed interaction between dopant incorporation and sputtering.

However, with results from many static simulations for varying diameters, a pseudo-dynamic study can be conceived. The sputter yield is dependent on the nanowire radius and the ion energy as shown in 2.5a. This relation is discussed in detail in the previous chapter ??. Like-

2 High Doping Concentrations in Nanowires

wise the incorporation efficacy plotted in 2.5b is also dependent on the nanowire radius and the ion energy. For a fixed diameter and increasing ion energy the efficacy is monotonically decreasing, as the probability of the ion to leave the nanostructure rises together with the ion range. For fixed ion energies the probability of an ion to stay in the nanostructure increases with increasing diameter, so that at first the efficacy also increases with increasing diameter. For large diameters this effect is overcompensated by a stronger dilution of the dopants in the volume of the nanowire which increases as the square of the diameter. This leads to a maximum in the incorporation efficacy at diameters around twice the ion range.

2.4 Discussion of relevant effects

Summarizing Discussion

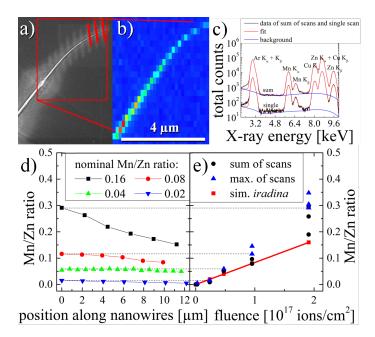


Figure 2.2: a) SEM image of a $175\,keV\,Mn^+$ irradiated ZnO nanowire on the carbon-foil of a Cu TEM grid after XRF investigation. The red lines indicate where the focused X-ray beam was scanned with a long integration time. b) Intensity map of the X-ray signal. c) Exemplary XRF-spectra of a single scanned line and for the sum of all the lines for the nanowire shown in a) and b). d) Mn/Zn ratio quantified with PyMCA for representative wires along the length of the nanowires for varying nominal concentrations. The corresponding data points in the plot of the concentration versus the irradiated ion fluence in e) are connected with a dashed line. The red data points and line in e) indicate the linear extrapolation to the nominal Mn/Zn ratio from iradina simulations. The black circles show the average ratio obtained by fitting to the sum of all scans, while the blue upturned triangles show the maximum ratio found for along the length of a nanowire.

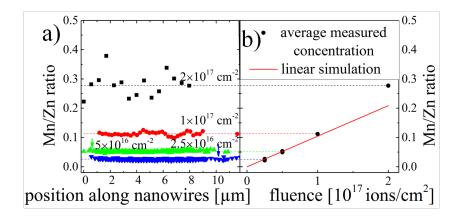
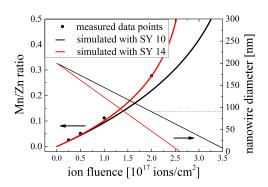


Figure 2.3: a) Mn/Zn ratio along the wire length for sparse nanowire samples irradiated with the indicated ion fluence of 175 keV Mn^+ . There is no concentration profile along the wire length. In b) the black circles show the average ratio obtained by fitting to the sum of all scans for the respective ion fluence. The red line in b) shows the linear extrapolation from iradina simulations.

Figure 2.4: Plot of the Mn/Zn ratio (left axis) versus the irradiated ion fluence for the measured nanowires, indicated by black circles, and two simulations with a constant sputter yield of 10 and 14, indicated by black and red lines respectively. The nanowire diameter (right axis) is also plotted against the fluence for both simulations. The dashed line at $90 \, nm$ marks the final radius of the data point corresponding to the highest irradiated fluence.



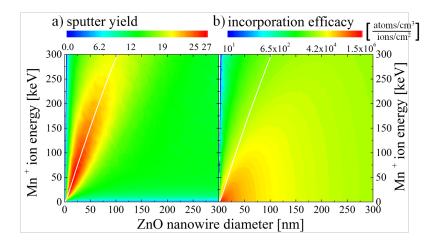


Figure 2.5: a) Sputter yield for the irradiation with Mn^+ of ZnO nanowires with varying diameters an ion energies. From the same simulations the dopant incorporation efficacy was determined and plotted in b). The white line in both plots indicates the ion range at the respective energy and 45° calculated with SRIM for Mn^+ in ZnO.

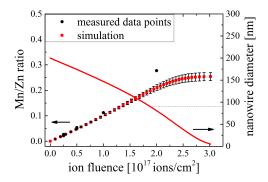


Figure 2.6: Illustration of the rotated, heated and tilted (RHT) stage for the ROMEO implanter. The rotating stage is tilted towards the ion beam by 45° and heated from the back by two light bulbs. The temperature is controlled by a reference measurement on a stationary disk similar to the rotating stage.

3 Summary and Outlook

check: Master Thesis Noack, Ogrisek, Conference proceding D. Sage, Rutherford, Nordlund

Bibliography

- [BMB+11] Christian Borschel, Maria E. Messing, Magnus T. Borgstrom, Waldomiro Paschoal, Jesper Wallentin, Sandeep Kumar, Kilian Mergenthaler, Knut Deppert, Carlo M. Canali, Hakan Pettersson, Lars Samuelson, and Carsten Ronning. A New Route toward Semiconductor Nanospintronics: Highly Mn-Doped GaAs Nanowires Realized by Ion-Implantation under Dynamic Annealing Conditions. Nano Letters, 11(9):3935—3940, September 2011. WOS:000294790200073.
- [Bor12] Christian Borschel. *Ion-Solid Interaction in Semiconductor Nanowires*. PhD thesis, University Jena, Jena, 2012.
- [JNP+14] A. Johannes, S. Noack, W. Paschoal, S. Kumar, D. Jacobsson, H. Pettersson, L. Samuelson, K. A. Dick, G. Martinez-Criado, M. Burghammer, and C. Ronning. Enhanced sputtering and incorporation of Mn in implanted GaAs and ZnO nanowires. *Journal of Physics D-Applied Physics*, 47(39):394003, October 2014. WOS:000341772000005.
- [KPJ+13] Sandeep Kumar, Waldomiro Paschoal, Andreas Johannes, Daniel Jacobsson, Christian Borschel, Anna Pertsova, Chih-Han Wang, Maw-Kuen Wu, Carlo M. Canali, Carsten Ronning, Lars Samuelson, and Håkan Pettersson. Magnetic Polarons and Large Negative Magnetoresistance in GaAs Nano-

Bibliography

- wires Implanted with Mn Ions. *Nano Letters*, 13(11):5079–5084, 2013.
- [PKB+12] Waldomiro Paschoal, Sandeep Kumar, Christian Borschel, Phillip Wu, Carlo M. Canali, Carsten Ronning, Lars Samuelson, and Hakan Pettersson. Hopping Conduction in Mn Ion-Implanted GaAs Nanowires. *Nano Letters*, 12(9):4838–4842, September 2012. WOS:000308576000069.
- [PKJ+14] W. Paschoal, Sandeep Kumar, D. Jacobsson, A. Johannes, V. Jain, C. M. Canali, A. Pertsova, C. Ronning, K. A. Dick, L. Samuelson, and H. Pettersson. Magnetoresistance in Mn ion-implanted GaAs:Zn nanowires. *Applied Physics Letters*, 104(15):153112, April 2014. WOS:000335145200060.