

# Integrated Thermo-mechanical Design of First Wall Components Under Evolving Chemistry and Microstructure During Fusion Reactor Operation

## ThermChem-FW

(An FES/Scidac-5 Center)

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# Project senior personnel

**UCLA**



Marian (Mat)

**Stony Brook University**



Trelewicz (Mat)

**W**



- Szlufarska (Mat)  
- Morgan (Mat)

**MIAMI**



Po (Mat)

**V**



Cereceda (Mat)



- Permann (Math)  
- Spencer (Math)

**OAK RIDGE**  
National Laboratory

- Bernholdt (CS)  
- Harb (Nucl)



- Cusentino (Mat)  
- Sargsyan (Math)



- Setyawan (Mat)

**INL**  
Idaho National Laboratory

**Pacific Northwest**  
NATIONAL LABORATORY

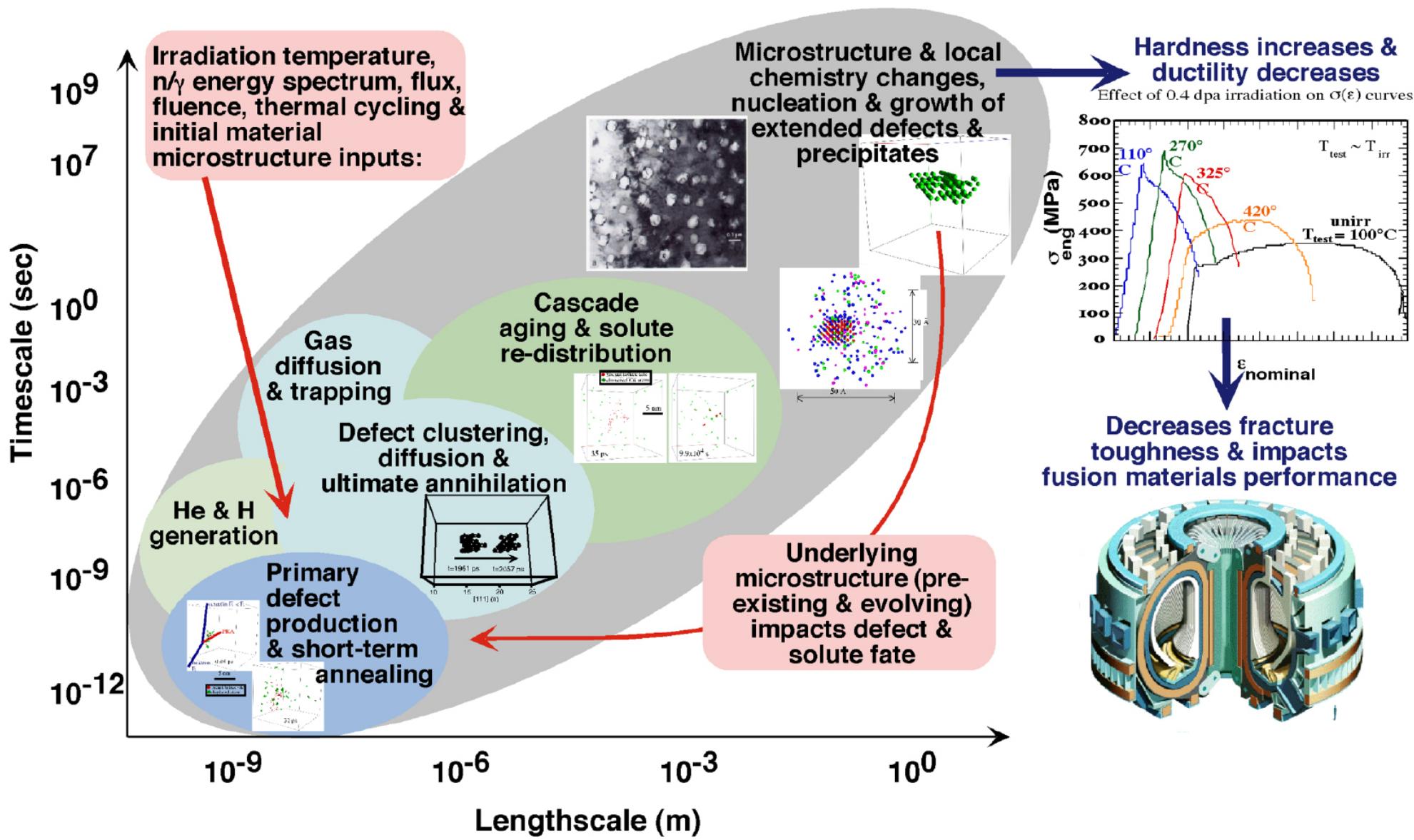
Plus :

- 6 Staff scientists (applied mathematicians, materials scientists, and computer scientists).
- 4 postdoctoral scholars
- 6 PhD students

# Multiscale analysis of irradiation damage processes in fusion materials

*Modeling paradigm:*  
sequential connection  
(parameter passing)  
across scales

*Objective:*  
derive ansatz-free,  
physics-based,  
predictive models of  
macroscopic behavior



# Solving the mathematical problem embodied by irradiation damage represents a computational grand challenge

Generalized mean-field cluster dynamics PDE based on classical nucleation theory

Golubov, Ghoniem,  
Odette, Barashev, Stoller,  
Xu and Wirth

$$\frac{\partial C_i}{\partial t} = \nabla \cdot (D_i \cdot \nabla C_i) + g_i + \left( \sum_j s_j C_j - s_i C_i \right) + \sum_j \left( \sum_k k_{jk} C_j C_k - k_{ij} C_i C_j \right)$$

Fickian diffusion

$$D_i = D_0^i \exp \left( -\frac{E_m^i}{kT} \right)$$

Crystal  
structure,  
chemistry

Species  
insertion

Irradiation  
conditions

Absorption at sinks and  
thermal dissociation

$$s_i^{\text{diss}} = 4\pi r_i^2 D_i \exp \left( -\frac{E_b^i}{kT} \right)$$
$$s_i^{\text{sinks}} = k_c^2 D_i$$

Microstructure

Production/destruction  
from binary reactions

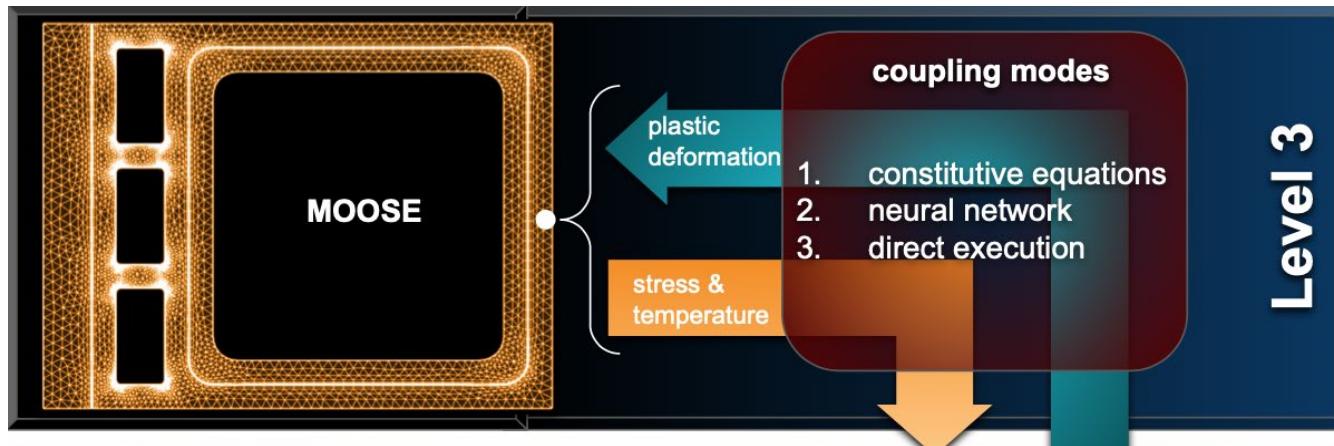
$$k_{ij} = 4\pi(r_i + r_j)(D_i + D_j)$$

Materials physics

All these coefficients give information about the physics of the process,  
the host material, and the microstructure

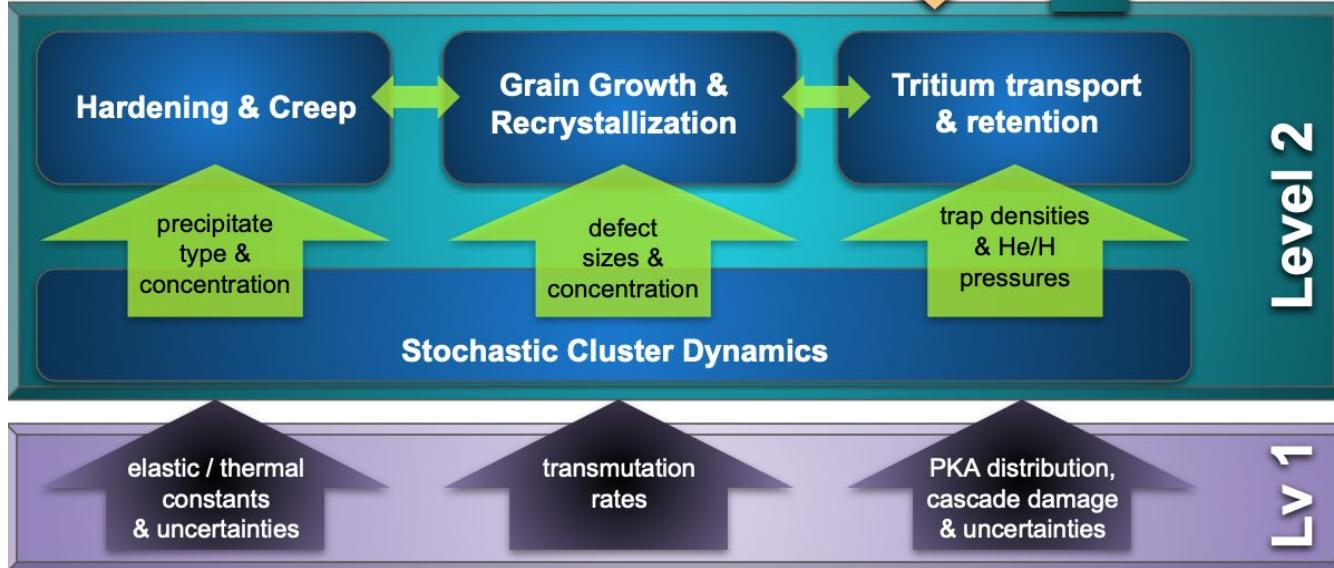
Damage accumulation models must incorporate all our latest knowledge on irradiation effects coming from MD,  
cascade damage, irradiation phenomena (RIS, RED, RIP), multispecies chemistry, etc. → this can lead to  $\sim 10^6$   
coupled equations

# The project has been structured as a connected multi-level workflow



**Level 3** involves time-dependent simulations (using the finite element code MOOSE) of the thermomechanical evolution of the full FW/B structure. These models will capture the entire geometric complexity and diversity of the multi-material FW/B and will furnish component lifetime prediction and design recommendations based on stress distributions and material property degradation.

*INL, ORNL, Miami*



**Level 2** operates at mesoscopic scales where microstructural evolution occurs, including irradiation hardening, irradiation creep and swelling, thermal fatigue, precipitation, recrystallization (grain growth), and tritium permeation and retention.

*UCLA, Miami, ORNL*

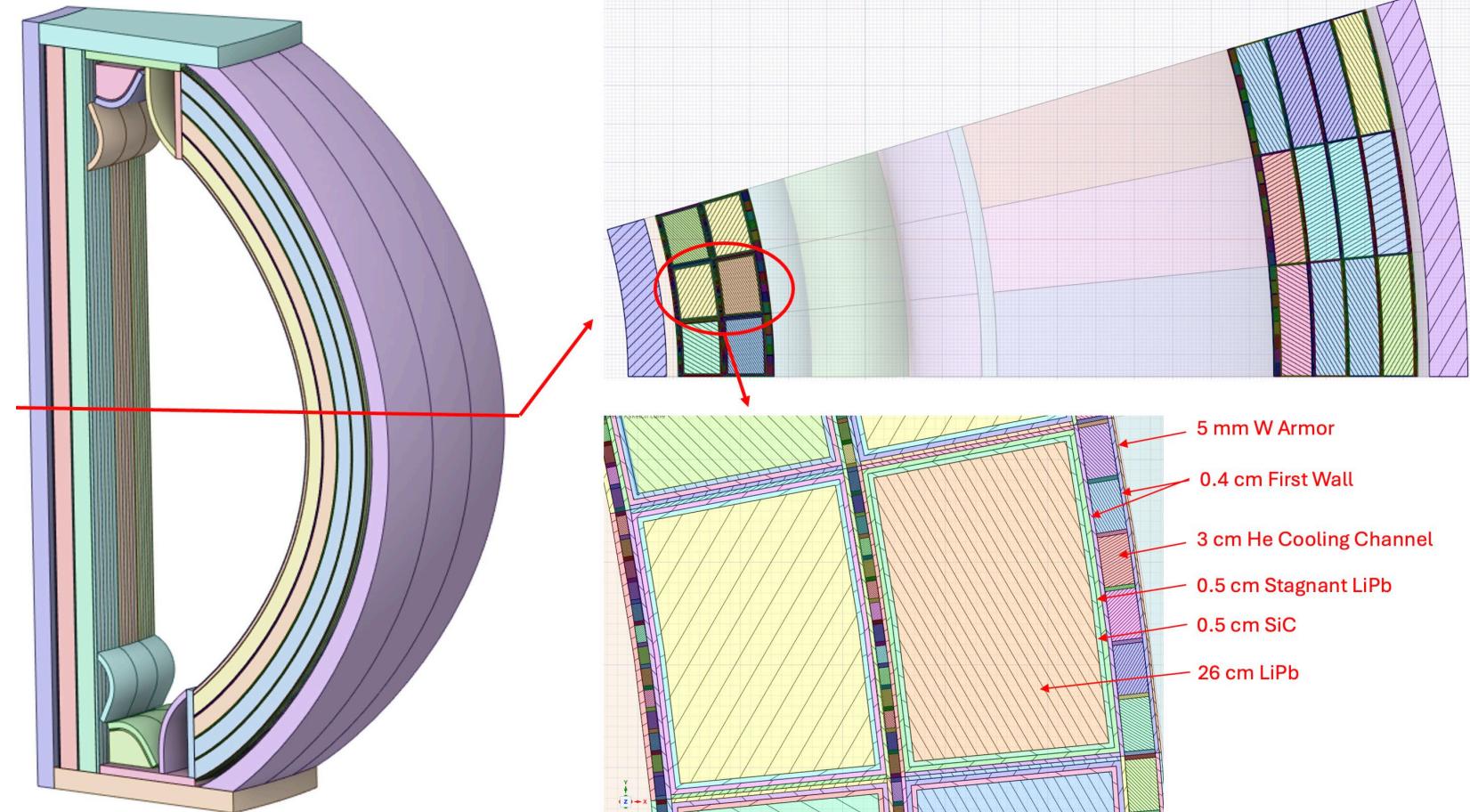
**Level 1** captures atomic-level physical processes representative of fusion reactor operation in the FW/B structure, such as (i) transmutation, primary knock-on atoms (PKA) distributions, (ii) cascade damage, (iii) interactions of defect clusters, and (iv) tritium uptake, trapping, and diffusion.

*Stony Brook, Wisconsin, Villanova, SNL*

# Model Description

Harb, Yakubov, Bernholdt

- Neutronics model contains He cooling channels in FW, SW, and breeding zone ribs, as well as SiC FCI.
- Material and neutron source definitions are taken from ARIES conceptual design (<https://doi.org/10.1016/j.fusengdes.2017.06.008>)
- CAD-based neutron transport was used via OpenMC v0.15.0 (<https://docs.openmc.org/en/stable/index.html>)
- Materials:  
Structures: W, SiC, RAFM steel  
He and LiPb colling/breeding channels



# Level 1: ML Potential Development Workflow

Cusentino, Sargsyan

## Model Form

- Energy of atom  $i$  expressed as a basis expansion over N-body ACE descriptors

$$E = \sum B\left(\begin{smallmatrix} N=1 \\ \circ \cdots \circ \end{smallmatrix}\right) + \sum B\left(\begin{smallmatrix} N=2 \\ \circ \cdots \circ \cdots \circ \end{smallmatrix}\right) + \sum B\left(\begin{smallmatrix} N=3 \\ \circ \cdots \circ \cdots \circ \cdots \circ \end{smallmatrix}\right) + \dots$$

The ACE descriptors are generalizable

## Regression Method

- $\beta$  vector fully describes an ACE potential
- Decouples MD speed from training set size

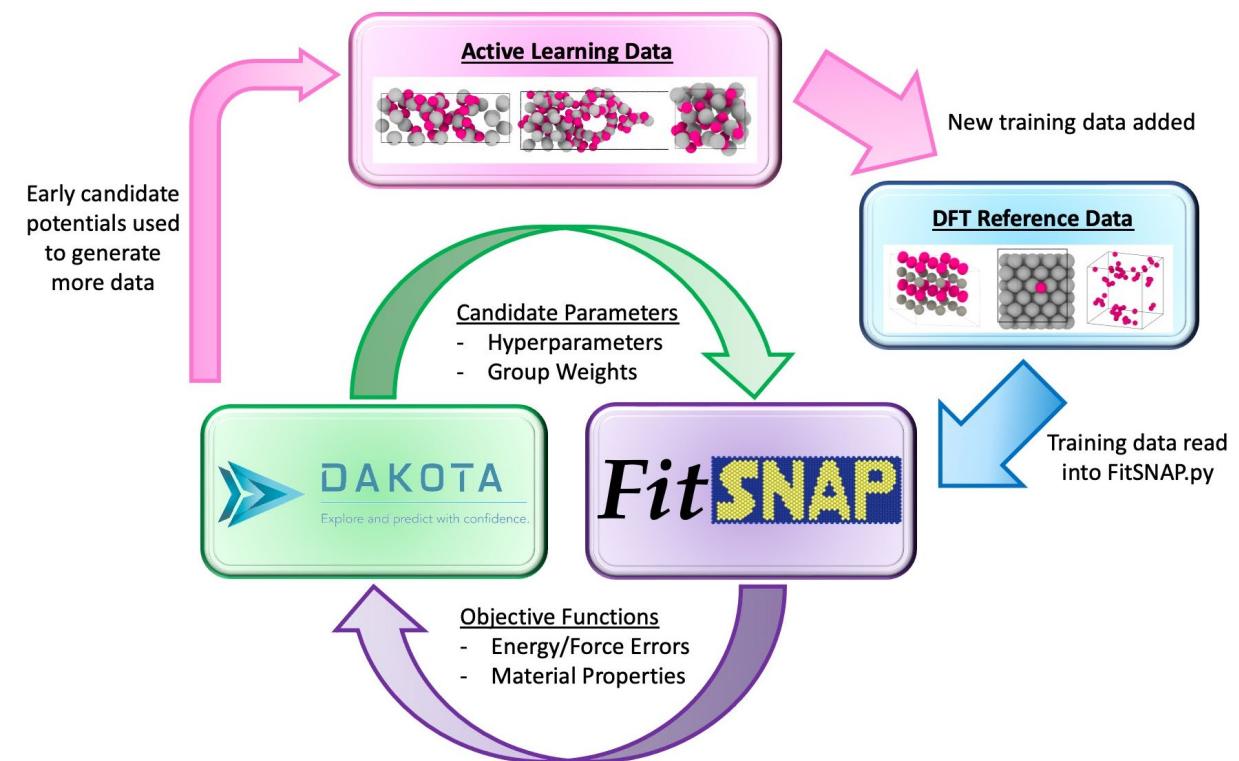
$$\min(||\mathbf{w} \cdot D\beta - T||^2 - \gamma_n ||\beta||^n)$$



## **Focus of this project:**

**W-Re-Os    SiC-Mg    Fe-Cr-Mn-W**

## ACE Development Workflow



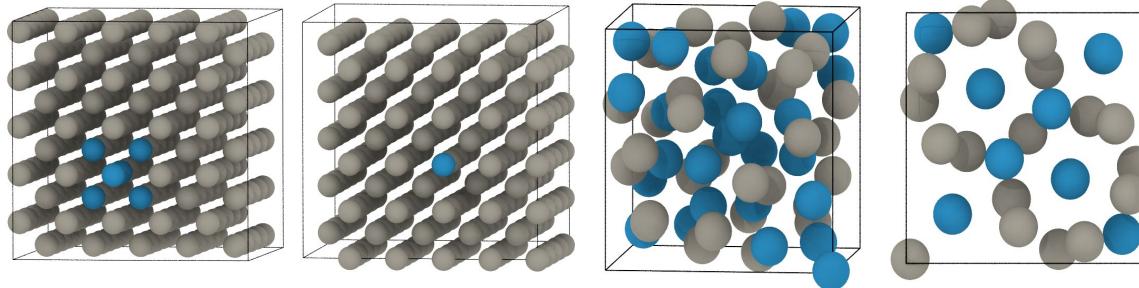
Code available: <https://github.com/FitSNAP/FitSNAP>

Rohskopf et al., (2023). FitSNAP: Atomistic machine learning with LAMMPS. Journal of Open Source Software, 8(84), 5118

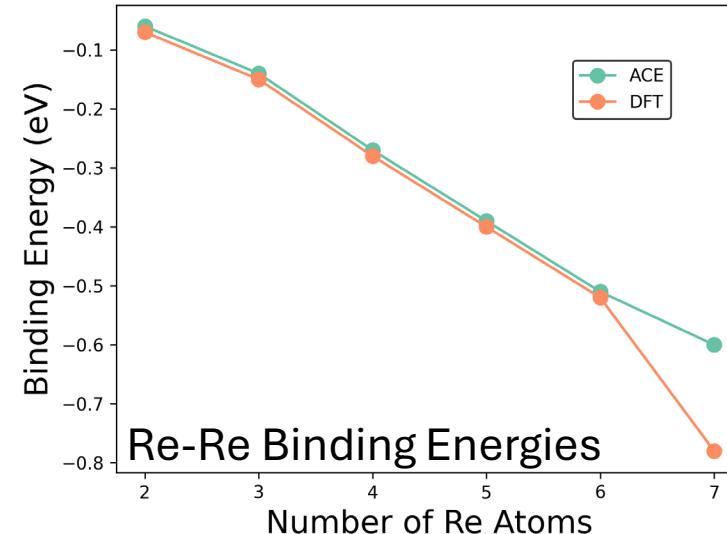
# Level 1: Development of ACE W-Re ML-IAP

Cusentino, Sargsyan

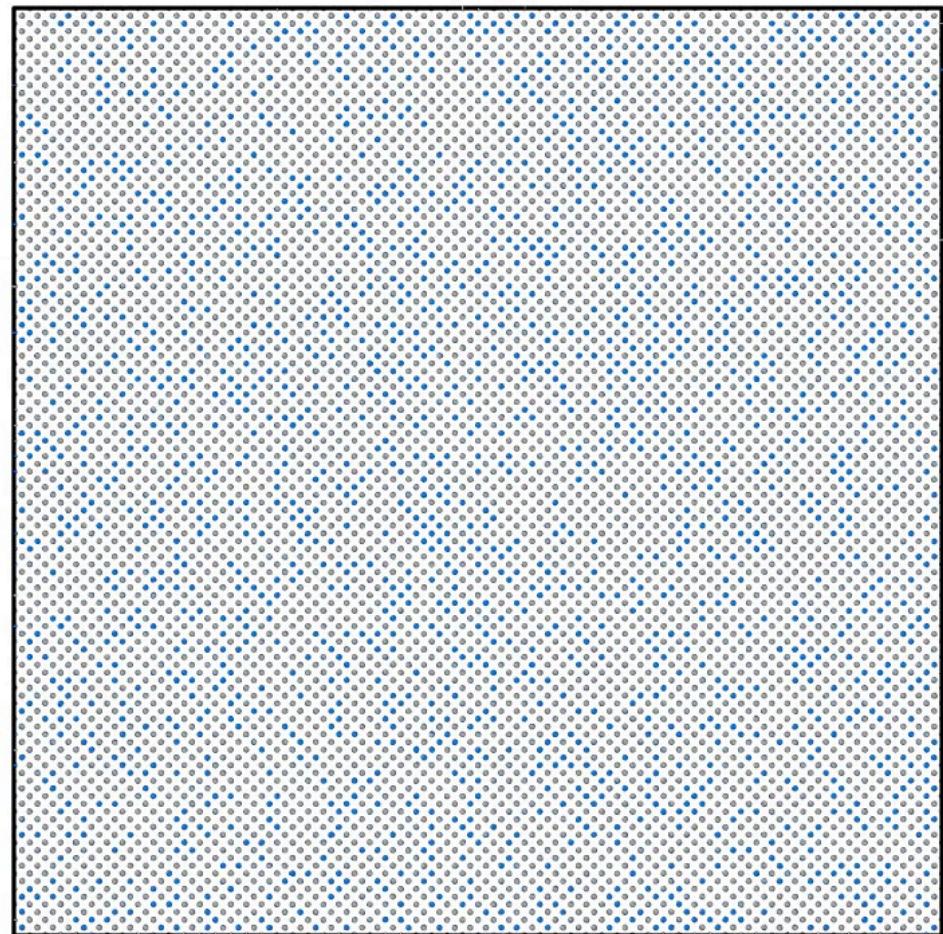
Training Data Includes Defects and  
Intermetallic Structures



ACE W-Re Reproduces Key Properties



ACE W-Re Stable for 10 KeV PKA Simulations



Migration Barriers (eV)	DFT	ACE
Vac	1.69	2.29
W SIA	0.003	0.02
Re	0.12	0.13

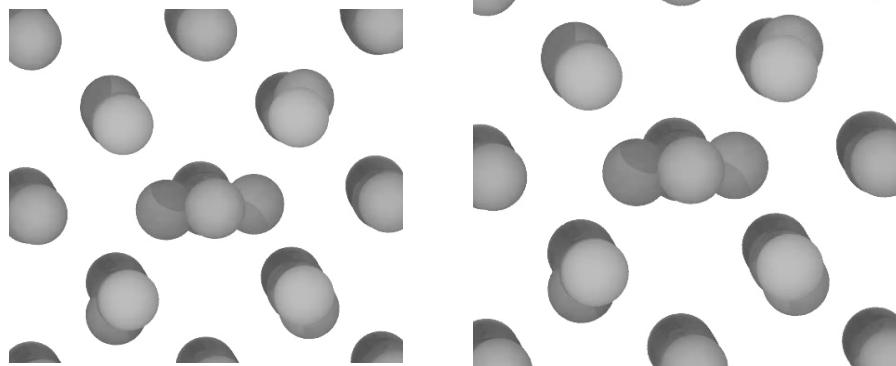
# Level 1: Development of Fe and SiC-Mg ACE ML-IAPs

Cusentino, Sargsyan

## Focus on Point Defect Properties

### NEB Barriers

110-111 Rotation



DFT: 0.67 ACE: 0.57

110-011 Rotation

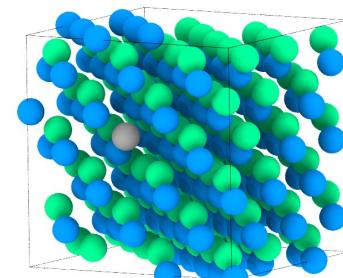
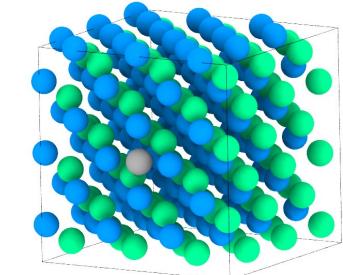
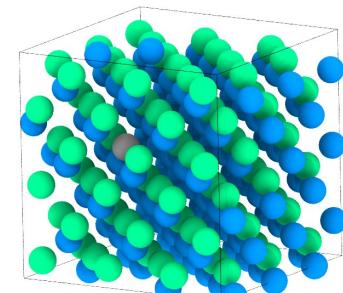
DFT: 0.58 ACE: 0.48

Vacancy Binding Energy (eV)	DFT	ACE
1 NN	0.13	0.13
2 NN	0.21	0.21
3 NN	-0.16	-0.09

## More Complex Defect Landscape

### Mg Defect Structures in 3C SiC

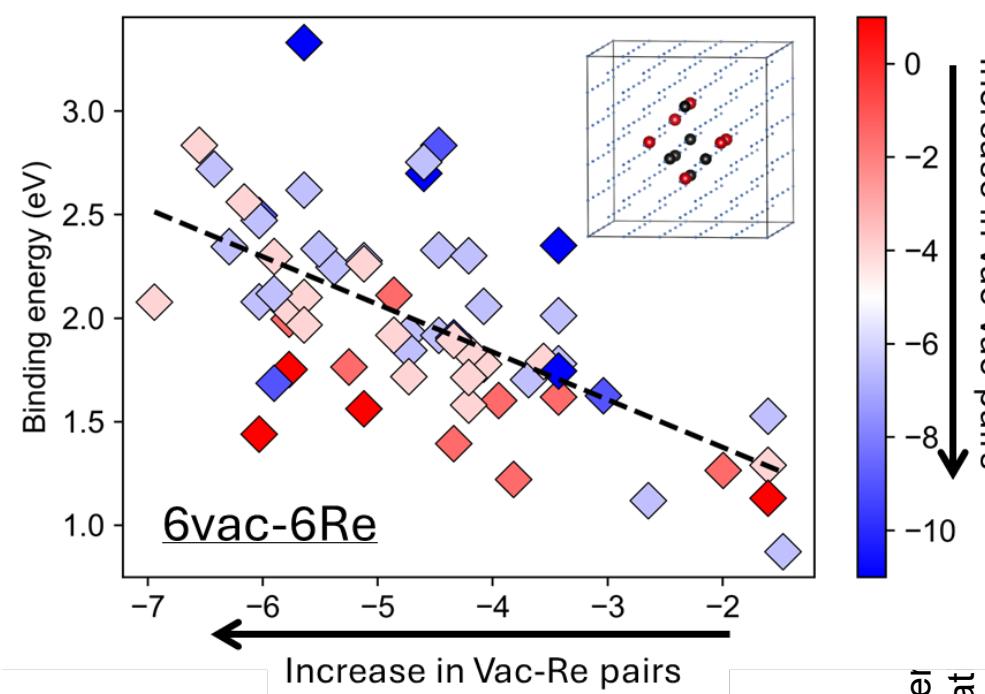
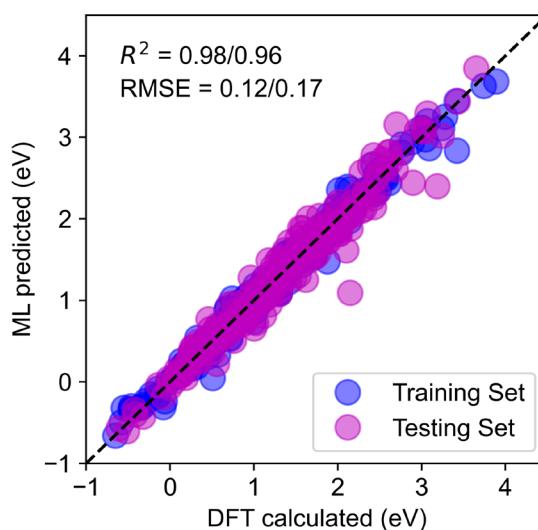
Mg Defect Formation Energies (eV)	DFT	ACE
$\text{Mg}_{\text{Si}}$	0.74	0.77
$\text{Mg}_{\text{Si}}-\text{V}_\text{C}$	1.52	1.55
$\text{Mg}_{\text{TC}}$	5.53	5.22
$\text{Mg}_{\text{TSi}}$	6.95	5.91
$\text{Mg}_{\text{Si}}-\text{V}_\text{Si}$	7.75	6.18
$\text{Mg}_\text{C}$	8.92	8.81
$\text{V}_\text{C}$	4.08	4.16
$\text{V}_\text{Si}$	5.26	4.65



# Application of the ML Model for Binding Energy

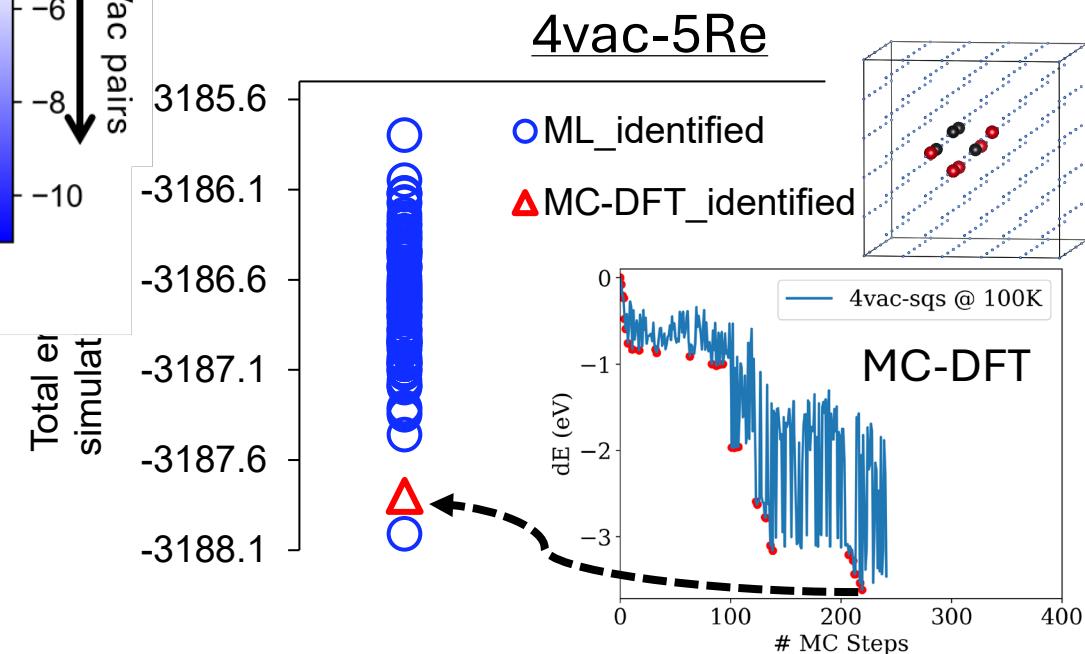
Manzoor, Jin, Thomas, Trelewicz

- Trained on 10 configurations for each composition
- Every random configuration is unique



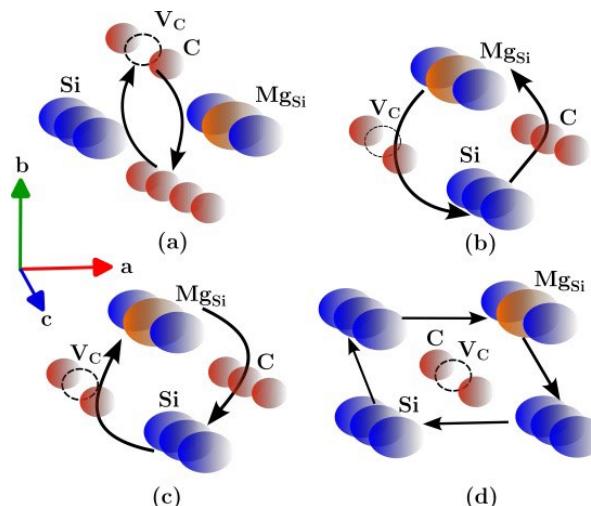
Identical binding energy trend as observed in DFT

- Low energy configurations are comparable between ML and DFT
- ML significantly less computationally expensive, i.e., 10 vs. 300 DFT calculations for a given composition



# Mg interactions with native defects in SiC

- 3C-SiC – a candidate structural material for the future fusion reactors.
- Radiation exposure causes damage and burns-in transmutants.
- Transmutants interact with native defects.
- 3C-SiC is a wide bandgap semiconductor – defects can be charged, and their energies depend on Fermi level  $\mu_F$ .



transmutant	burn-in rate (appm/efpy)	after 6.5 efpy (at. %)
He	6384	4.2
H	2307	1.5
Mg	1630	1.1
Be	632	0.4
Al	469	0.3
P	146	0.1

[1] R.H. Jones, [...], W. J. Weber, *Journal of Nuclear Materials* **307–311**, 1057 (2002)

[2] Rafi Ullah, Dane Morgan, Izabela Szlufarska, *J. Phys. D: Appl. Phys.* **58** (2025) 265302

# Level 2: Calculation of material properties using mesoscale simulations

Marian, Szlufarska

Ray-tracing Monte Carlo model of electronic dynamics to calculate the **thermal conductivity** of tungsten and its alloys under changing microstructure

Computational cost and error estimation

Calculated vs experimental in pure W

Calculated vs experimental in W-Re

Calculated thermal and concentration dependence



ThermChem-FW

A research project supported by the Scientific Discovery through Advanced Computing (SciDAC) program of the U.S. Department of Energy

# Level 2: Simulations of helium-defect interactions in irradiated RAFM steel

Marian, Po

Cluster dynamics simulations using grouping methods to reduce the number of ODEs from  $10^8$  to  $10^4$ , and reach dpas  $\sim 50$  dpa.

Inclusion of heterogeneous nucleation of bubbles in irradiated RAFM steels

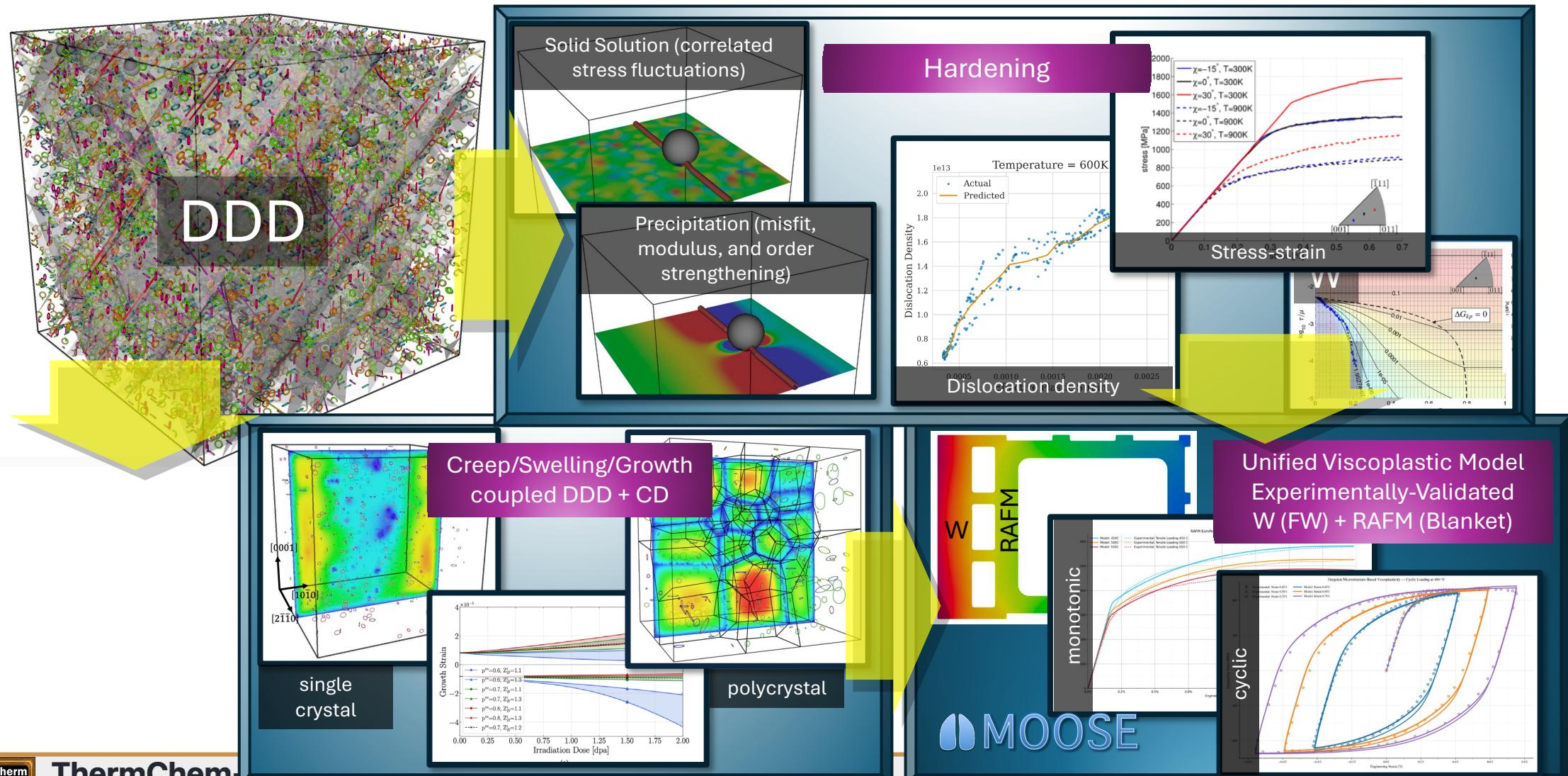
Equation of state for V-He clusters (bubbles) used to obtain the critical He-to-vacancy ratio

Using grouping methods reduces the number of ODEs to tractable numbers, allowing reaching 50 dpa and enabling comparison with experiments



# Level 3: Integration of mesoscale models with engineering design codes

Po, Spencer



# Collaboration resources

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- The project mailing list ([thermchem-fw@email.ornl.gov](mailto:thermchem-fw@email.ornl.gov))
- The Box workspace we use for internal file sharing  
(<https://ucla.box.com/s/o2unb7df1ep12g1lefwdwsm7e0fmttf49>)
- The Slack workspace we use for informal communications  
(<https://thermchem-fwscidac.slack.com/home>)
- The GitHub team(s) associated with our GitHub organization  
(<https://github.com/ThermChem-FW>)
- Our website: <https://thermchem-fw.github.io/>
  - Which uses the Jekyll static site generation tool and is hosted in this repository:  
<https://github.com/ThermChem-FW/thermchem-fw.github.io>
  - Information on accessing various computing facilities here: [Computer Access.boxnote](#)



# Nuclear Hearing

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- Nuclear heating was obtained over a 1x1x1-cm cartesian mesh.
- Figures below show nuclear heating [W/cc] - at vertical slice (left) and horizontal (middle) slices – and neutron flux [n/cm<sup>2</sup>/s] (right).
- Cylindrical mesh could be considered to align better with the structural zones.

