

DISSERTATION

IMAGING OF SINGLE BARIUM ATOMS IN SOLID XENON FOR BARIUM TAGGING IN THE  
nEXO NEUTRINOLESS DOUBLE BETA DECAY EXPERIMENT

Submitted by

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In partial fulfillment of the requirements

For the Degree of Doctor of Philosophy

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Fall 2015

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## ABSTRACT

The nEXO experiment is designed to search for zero-neutrino double beta decay of the isotope  $^{136}\text{Xe}$ , in order to better understand the nature of neutrinos. Since the daughter of this decay is barium ( $^{136}\text{Ba}$ ), detecting the presence of  $^{136}\text{Ba}$  at a decay site (called "barium tagging") provides an additional discriminator to reject backgrounds in the search for this decay. This would involve detecting a single barium ion from within a macroscopic volume of liquid xenon. One proposed barium tagging method is to trap the barium ion in solid xenon (SXe) at the end of a cold probe, and then detect the ion by its fluorescence in the solid xenon. William M. Fairbank Jr.'s group at Colorado State University has been working toward this goal with steady success for some time. In this thesis, I demonstrate successful detection of neutral Ba in SXe down to the single atom level, after deposition from vacuum.

## ACKNOWLEDGEMENTS

Bill, Chris for all the help in the lab and building things, Shon and Brian for creating the system and training me (Shon) and for the preliminary results, Adam, Cesar, Kendy.

If you want the Leif thing, uncomment it in csuthesis.cls. It says something like "this dissertation is typeset in ... designed by Leif Anderson.

Jamie and LEW, and other fam.

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## CHAPTER 1

### INTRODUCTION

Neutrinos have been at the forefront of discovery since their prediction by W. Pauli, who proposed in 1930 the existence of a neutral, unobserved particle to explain the apparent violation of energy conservation in beta decay [1]. He admitted that neutrinos (then deemed “neutrons” – what we now know as neutrons had not been discovered yet either) should be difficult to observe experimentally, but also that it seemed unlikely that they would never have been noticed before. As it turns out, they are much more difficult to observe than he predicted; they will not be noticed without extreme experimental techniques.

A theory formulated in 1933 by E. Fermi for beta decay [2], including the neutrino, would be the beginnings of weak theory, and the development of the very successful Standard Model (SM) of particle physics. However, the SM assumes that neutrinos are massless. The discovery of non-zero neutrino mass via neutrino oscillations in the late 1990s [3] shows that there is more new physics to be discovered – a path to follow in the further unification of physical theory.

Neutrinos are extremely difficult to study, but the rewards are profound. The favored See-saw Mechanism is a theory which predicts the extreme lightness of neutrinos, while also predicting very heavy neutrinos. The existence of heavy neutrinos in the high-energy environment of the early universe, coupled with the possibility of the violation of CP conservation which could slightly favor the production of matter vs. anti-matter in decays of these heavy neutrinos, could explain why the universe exists as we know it. [4]

If this theory is correct, then neutrinos would be described by the Majorana formulation rather than Dirac, and the process of neutrinoless double beta decay ( $0\nu\beta\beta$ ) would be allowed. Observation of  $0\nu\beta\beta$  would simultaneously demonstrate that neutrinos are Majorana particles, as well as give a measurement of the absolute mass itself [5]. This chapter outlines the current theory for neutrinos, and then describes the  $0\nu\beta\beta$  experiments EXO-200 and nEXO, in order to motivate barium tagging for nEXO.

### 1.1. NEUTRINOS

Neutrinos are chargeless leptons which only interact via the weak force (and gravity). There are three known “flavors” of neutrinos, each corresponding to one of the three known leptons:  $\nu_e$ ,  $\nu_\mu$ , and  $\nu_\tau$ . These are the eigenstates in the basis of the weak force, so they are the states in which a neutrino will interact via the weak force.

**1.1.1. NEUTRINO OSCILLATION AND MASS.** The postulate that neutrinos have an energy basis which is different from the flavor basis predicts the phenomenon of oscillation – that the time evolution of an initially pure flavor state (as a neutrino will be produced) will result in a time-dependent probability of measuring the other two flavors as well.

The very small mass of a neutrino (assumed zero in the SM), specifically relative to its momentum, lets one write its Hamiltonian in terms of mass squared differences  $\Delta m_{ij}^2 = m_i^2 - m_j^2$ , where  $i,j = 1,2,3$ , referring to what we then call mass states. The mass basis is really the energy basis with the small mass approximation, along with dropping some constant terms in the Hamiltonian (which do not affect time evolution). Writing the time evolution in terms of mass squared differences means that neutrino oscillation experiments can produce measurements of these differences. In fact, the discovery of neutrino oscillation was the first (and only, so far) demonstration that neutrinos have a non-zero mass. Without

neutrino mass (particularly without differences between the masses of the mass states), neutrinos would not oscillate.

Neutrino oscillation experiments also provide measurements on the amount of mixing between the flavor basis and the mass basis. We define the mixing between them by a rotation in terms of three mixing angles,  $\theta_{12}$ ,  $\theta_{23}$ , and  $\theta_{13}$ . Transformation between the flavor and mass bases is done with the following unitary matrix, called the Pontecorvo–Maki-Nakagawa-Sakata (PMNS) matrix:

$$(1) \quad U = \begin{pmatrix} 1 & 0 & 0 \\ 0 & c_{23} & s_{23} \\ 0 & -s_{23} & c_{23} \end{pmatrix} \begin{pmatrix} c_{13} & 0 & s_{13}e^{-i\delta} \\ 0 & 1 & 0 \\ -s_{13}e^{i\delta} & 0 & c_{13} \end{pmatrix} \begin{pmatrix} c_{12} & s_{12} & 0 \\ -s_{12} & c_{12} & 0 \\ 0 & 0 & 1 \end{pmatrix} \begin{pmatrix} 1 & 0 & 0 \\ 0 & e^{i\alpha_1/2} & 0 \\ 0 & 0 & e^{i\alpha_2/2} \end{pmatrix}$$

$$= \begin{pmatrix} c_{12}c_{13} & s_{12}c_{13} & s_{13}e^{-i\delta} \\ -s_{12}c_{23} - c_{12}s_{23}s_{13}e^{i\delta} & c_{12}c_{23} - s_{12}s_{23}s_{13}e^{i\delta} & s_{23}c_{13} \\ s_{12}s_{23} - c_{12}c_{23}s_{13}e^{i\delta} & -c_{12}s_{23} - s_{12}c_{23}s_{13}e^{i\delta} & c_{23}c_{13} \end{pmatrix} \begin{pmatrix} 1 & 0 & 0 \\ 0 & e^{i\alpha_1/2} & 0 \\ 0 & 0 & e^{i\alpha_2/2} \end{pmatrix}$$

where  $c_{ij} = \cos \theta_{ij}$  and  $s_{ij} = \sin \theta_{ij}$ .  $\delta$  is a phase factor related to lepton CP violation, and  $\alpha_i$  are Majorana phases.

Studying oscillations of neutrinos from different kinds of sources, with different energies and path lengths, can isolate sensitivities to the different parameters. For example, the study of solar neutrinos (neutrinos emanating from nuclear fusion reactions in the core of the sun) provides sensitivity to  $\theta_{12}$  and  $\Delta m_{12}^2$ . The oscillation parameters so far measured shown in Table 1.1:

TABLE 1.1. Best-fit values for neutrino oscillation parameters, from a global fit to oscillation experiment data. Parameters which depend on the mass hierarchy have values for NH (IH). The atmospheric parameter  $\Delta m^2$  is defined as  $\Delta m^2 = \Delta m^2 = \Delta m_{31}^2 - \Delta m_{21}^2/2 > 0$  ( $\Delta m^2 = \Delta m_{32}^2 + \Delta m_{21}^2/2 < 0$ ). [6]

Parameter	Measurement ( $\pm 1\sigma$ )
$\Delta m_{12}^2$	$7.54^{+0.26}_{-0.22} 10^{-5} \text{ eV}^2$
$ \Delta m^2 $	$2.43 \pm 0.06$ ( $2.38 \pm 0.06$ ) $10^{-3} \text{ eV}^2$
$\sin^2 \theta_{12}$	$0.308 \pm 0.017$
$\sin^2 \theta_{23}$	$4.37^{+0.033}_{-0.023}$ ( $4.55^{+0.039}_{-0.031}$ )
$\sin^2 \theta_{13}$	$0.0234^{+0.0020}_{-0.0019}$ ( $0.0240^{+0.0019}_{-0.0022}$ )
$\delta/\pi$ ( $2\sigma$ range)	$1.39^{+0.38}_{-0.27}$ ( $1.31^{+0.29}_{-0.33}$ )

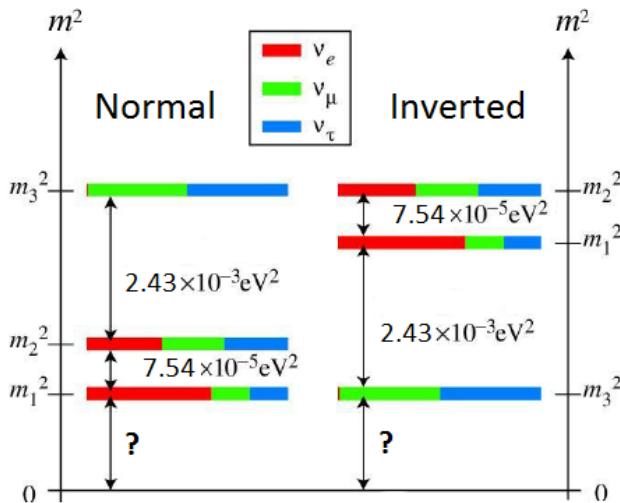


FIGURE 1.1. The two possible hierarchies of neutrino masses. The colors depict the mixing between the mass and flavor bases.

Note that only the absolute value of the atmospheric neutrino oscillation parameter  $\Delta m^2$  is known. As a consequence, there are two possibilities for the hierarchy of the three neutrino masses. These are called the Normal and Inverted Hierarchies, as shown in Fig. 1.1. The correct mass hierarchy remains unknown, but next-generation neutrino experiments, possibly including nEXO, will be able to discern this.

Neutrino oscillation demonstrates that neutrinos have non-zero mass, and though oscillation experiments measure the mass squared differences, we still do not have a measurement of the absolute masses of the three neutrinos. Cosmology can put limits on the sum of the



FIGURE 1.2. Two-neutrino (left) and neutrinoless (right) double beta decay.

three neutrino masses. The Planck collaboration reports an upper bound on this sum at  $\sum_i m_i < 0.23$  eV [7]. The KATRIN experiment will search for absolute neutrino mass with limiting capability of  $m_{\bar{\nu}_e} < 0.2$  eV (90% CL) by measuring the spectrum of tritium beta decay near the Q-value, searching for deviation from the spectrum with zero neutrino mass [8].

## 1.2. DOUBLE BETA DECAY

Double beta decay is the simultaneous decay of two neutrons in a nucleus into two protons and two electrons. Two-neutrino double beta decay ( $2\nu\beta\beta$ ), shown in Fig. 1.2(left), is allowed by the Standard Model and has been observed in eleven isotopes with half-lives between  $10^{19}$  and  $10^{21}$  years. Similar to beta decay, a neutrino accompanies each electron in this decay, broadening the spectrum of the summed electron energy. This is a second-order process, making it a rare decay, and requiring low backgrounds to measure.

$0\nu\beta\beta$ , shown in Fig. 1.2(right), is a postulated mode of double beta decay. In this case, the neutrino is exchanged as a virtual particle (which would require that it is a Majorana

particle), and there are no neutrinos in the final products. If discovered, not only would neutrinos be determined Majorana particles, but their absolute mass could also be measured as it relates to the  $0\nu\beta\beta$  half-life according to Eqn. 2:

$$(2) \quad T_{1/2}^{0\nu} = (G^{0\nu}(Q, Z)|M^{0\nu}|^2 \langle m_\nu \rangle^2)^{-1}$$

where  $T_{1/2}^{0\nu}$  is the  $0\nu\beta\beta$  half-life,  $G^{0\nu}$  is a known phase space factor, and  $M^{0\nu}$  is a model-dependent nuclear matrix element. The effective electron neutrino mass  $\langle m_\nu \rangle$  is the expectation value of the mass for a pure electron neutrino:

$$(3) \quad \langle m_\nu \rangle = \sum_i U_{ei}^2 m_i.$$

The sum of the energies of the emitted electrons in double beta decay will serve as the distinction between the two-neutrino and zero-neutrino modes, shown in Fig. 1.3. In the two-neutrino mode, the total decay energy is shared probabilistically between the electrons and the neutrinos (the nucleus recoil energy is negligible), resulting in a broad distribution in the summed electron energy. (Recall the similarly broad electron energy in single beta decay, which ultimately led to discovery of the neutrino involved.) But in the zero-neutrino mode, all of the decay energy is carried away by the two electrons, resulting in only a single allowed value for the summed electron energy – a peak in the summed electron energy spectrum at the Q-value.



FIGURE 1.3. Conceptual two-neutrino (blue) and zero-neutrino (red) double beta decay spectra.

The rarity of double beta decay requires very low backgrounds, especially around the Q-value for the  $0\nu\beta\beta$  search. The next sections describe the experiments EXO-200 and its next-generation successor, nEXO.

### 1.3. ENRICHED XENON OBSERVATORY

EXO-200 and nEXO (EXO standing for Enriched Xenon Observatory) are a progression of two experiments, each a LXe time projection chamber (TPC) designed to study the double beta decay of the isotope  $^{136}\text{Xe}$ , and ultimately to search for the zero-neutrino mode. Xe is fairly unique among the double beta decay isotopes in that it can be studied in a gas or liquid TPC instead of solid crystals or foils. The 3D event position reconstruction abilities of a TPC have advantages in background reduction, described in section 1.3.1. Purification of Xe is straightforward and can be done continuously in a detector. LXe is transparent, and produces substantial ionization and scintillation at 178 nm when energy is deposited in the LXe [9]. A liquid TPC approach also offers the opportunity to identify, or “tag”, the

daughter Ba<sup>++</sup> at the site of the double beta decay event, which would provide a background-free identification of  $0\nu\beta\beta$  [10]. Barium tagging is the focus of our group at CSU and is the subject of this thesis. The following sections describe the EXO-200 experiment, as well as nEXO, the next-generation tonne-scale LXe TPC which is now in the design stages. EXO-200 does not have barium tagging implemented, but it is hoped that nEXO will.

1.3.1. EXO-200. EXO-200 has been operational since 2011. It is a TPC using 110 kg of active LXe enriched to  $80.672\% \pm 0.14\%$   $^{136}\text{Xe}$  [9], designed to probe Majorana neutrino masses down to around 109-135 meV (the range covers different matrix element calculations) [11], and is located about half a mile underground in the Waste Isolation Pilot Plant (WIPP) near Carlsbad, NM. This mine is in a salt basin, which contains lower levels of Uranium and Thorium than a more typical mine in rock.

A schematic of the TPC in the class 100 cleanroom is shown in Fig. 1.4. Several layers of lead wall surround the copper cryostat, which is filled with HFE-7000, a cryogenic fluid which keeps the TPC cooled to LXe temperatures, as well as aids in shielding. The TPC vessel is made of low-radioactivity copper, and is kept as thin as possible to minimize backgrounds. Scintillating panels on the outside of the cleanroom provide muon veto.

A cut-away view of the EXO-200 detector is shown in Fig. 1.5. It is two mirrored TPCs which share a cathode. The detection planes are a combination of ionized charge induction/collection wires and large area avalanche photodiodes (LAAPDs) which detect scintillation light [12].



FIGURE 1.4. Drawing of cleanroom laboratory in the WIPP drift.

A photograph of the detection plane is shown in Fig. 1.6, and a schematic of event detection is shown in Fig. 1.7. When a double beta decay event occurs in the LXe, the energetic electrons ionize many surrounding Xe atoms, as well as produce a scintillation signal. Scintillation light provides a time stamp for the event as well as an energy measurement.

The cathode is set to -8 kV, providing an electric field of 374 V/cm across the 20 cm drift length of each TPC. Ionized electrons drift from the decay site, first passing the v-wires, which receive an induction signal, and are then collected by the u-wires, which are set at a 60° angle from the v-wires. An electric field of 778 V/cm between the u- and v-wires ensures 100% v-wire transparency. The charge collection provides an energy measurement complimentary to that of the scintillation. Together, the u- and v-wires give an x/y position measurement for the event. The time between the initial scintillation detection and the charge collection give a



FIGURE 1.5. EXO-200 TPC cutaway [9].

$z$  position, and a 3D position can be reconstructed for the event. [9] Additional scintillation resulting from recombination of ionized electrons leads to a well-known microscopic anti-correlation between ionization and scintillation signals [13]. Applying this to the combination of those signals improves the energy resolution. This correction defines a combined energy axis, called the rotated energy. Energy resolution is important in a  $0\nu\beta\beta$  search, as it distinguishes those events from  $2\nu\beta\beta$  events in the tail of their spectrum.



FIGURE 1.6. View of the detection plane in one of the two EXO-200 TPCs.



FIGURE 1.7. EXO-200 event detection.

Having a reconstructed 3D event position is important in several ways. Firstly, position-based corrections on scintillation and charge collection can be applied. For charge, electronegative impurities in the LXe will absorb the drifting charge, requiring a drift-length (z-position) correction. High purity levels, measured in terms of electron lifetime, of  $2\text{--}3 \times 10^3 \mu\text{s}$  are maintained in EXO-200, resulting in a small correction of few % for maximal drift lengths. For scintillation, a 3D correction is applied, as some regions have more efficient light collection by the LAAPDs. A 3D position also allows a fiducial volume to be defined. The effect of radio-impurities on detector surfaces, e.g. radon daughters collecting on the cathode, is mitigated by defining a stand-off distance required for events used in analysis. Finally, 3D reconstruction allows the distinction between single-site (SS) and multi-site (MS) events. A MS event is one where two spatially separated events occur in the same  $2048\text{-}\mu\text{s}$  time window. These are mostly caused by gamma rays interacting in the LXe, which can Compton-scatter several times. Rejecting MS events strongly separates gamma events from double beta decay events. [9] Of course, barium tagging will also require a 3D reconstructed position in nEXO.

The final data set is fit using a combination of probability distribution functions (PDFs) for  $0\nu\beta\beta$ ,  $2\nu\beta\beta$ , and all possible backgrounds. Fits to the final energy spectrum data for (a) SS events, and (b) MS events are shown in Fig. 1.8 for the most recent  $0\nu\beta\beta$  search analysis. The green bands beneath each plot show the residuals vs. energy. The  $2\nu\beta\beta$  spectrum, in gray, dominates the backgrounds in the SS spectrum. The vertical red lines in the SS spectrum outline the  $\pm 2\sigma$  region of interest around the Q-value, where the  $0\nu\beta\beta$  peak will lie. The insets are a zoom into this region. The fit value for  $0\nu\beta\beta$  in this dataset is non-zero, but it is consistent with the null hypothesis at  $1.2\sigma$ . This sets an upper limit on the



FIGURE 1.8. EXO-200 energy spectrum from  $0\nu\beta\beta$  search analysis [14].

$0\nu\beta\beta$  half-life at  $T_{1/2}^{0\nu\beta\beta} < 1.1 \times 10^{25}$  yr (90% CL), which corresponds to  $\langle m_{\nu_e} \rangle < 190\text{-}450$  meV depending on nuclear matrix element calculations [14]. EXO-200 reports the  $2\nu\beta\beta$  half-life measurement in [9] of  $T_{1/2}^{2\nu\beta\beta} = 2.165 \pm 0.016(\text{stat}) \pm 0.059(\text{sys}) \times 10^{21}$  yr. This is consistent with previous measurements by EXO-200 in 2011 [15] and KamLAND-ZEN in 2012 [16].

1.3.2. nEXO. The next-generation successor to EXO-200 is nEXO, a tonne-scale LXe TPC which will probe Majorana neutrino masses down to the 10 meV scale. The sensitivity projections for nEXO are shown in Fig. 1.9, along with those of EXO-200. nEXO will reach phase space where the two possible mass hierarchies begin to split; if nEXO successfully



FIGURE 1.9. nEXO projected sensitivity to the Majorana neutrino mass vs. the minimum of the three masses  $m_{min}$ . [check for updated version](#)

observes  $0\nu\beta\beta$  in these regions, it may be able to also determine the mass hierarchy. Barium tagging will push the sensitivity further into the region allowed only by the normal hierarchy [ref for this?].

A schematic of the experimental setup is shown in Fig. 1.10 in one of the possible locations for nEXO, the SNOLAB cryopit. Similar to EXO-200, the copper-housed TPC will be submerged in HFE fluid, inside a copper cryostat. The cryostat will be insulated and submerged in a large volume of water shielding, in which photo-multiplier tubes could

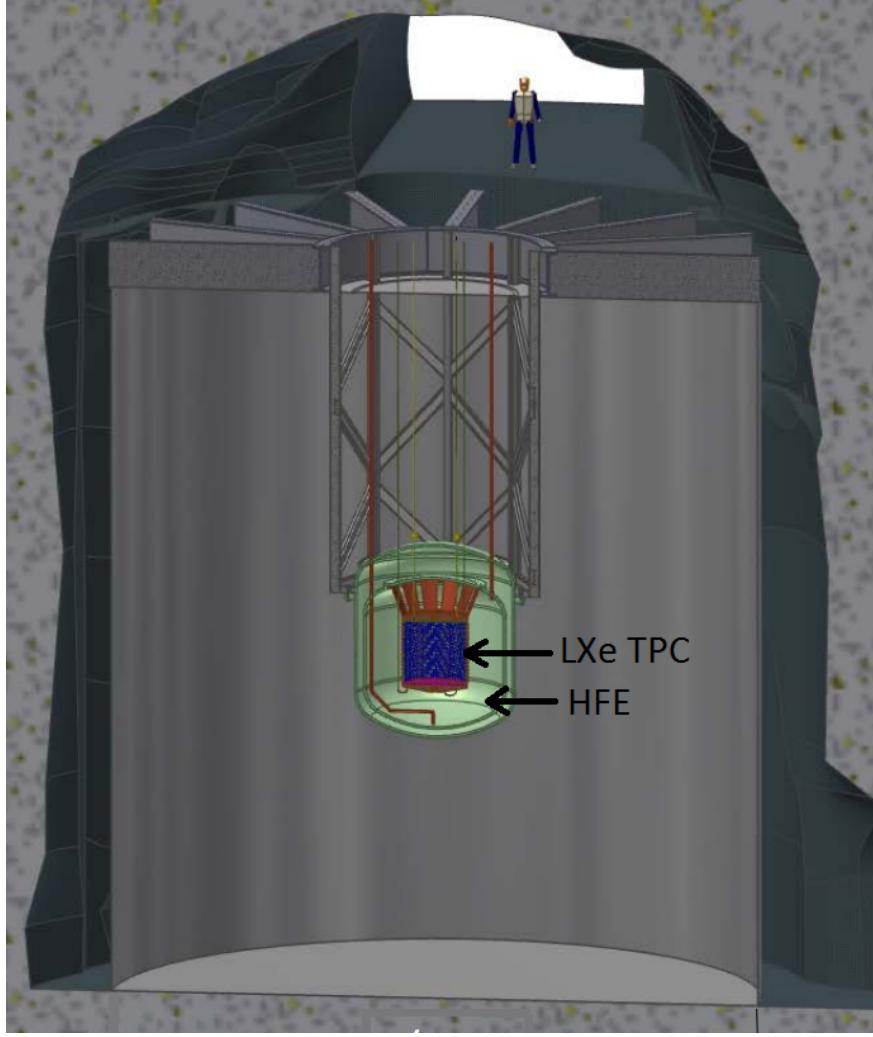


FIGURE 1.10. nEXO TPC in the SNOLAB cryopit.

provide muon veto by observing Cherenkov radiation. nEXO will be a single TPC with charge and light readouts at opposing ends of the TPC. Rather than wires, nEXO will use tile electrodes for charge readout, shown in Fig. 1.11.

**1.3.3. BARIUM TAGGING.** The backgrounds observed around the Q-value in EXO-200 can be expected to scale up with the size of nEXO. But with the veto power of barium tagging, nEXO's sensitivity to  $0\nu\beta\beta$  scales with the total  $^{136}\text{Xe}$  mass in the detector, vs. the square root of that mass without barium tagging.



FIGURE 1.11. nEXO readout.

Several possible barium tagging techniques have been proposed. Perhaps the most natural concept is to direct one or more lasers at the decay site to induce fluorescence of the barium daughter. This technique was explored thoroughly by Fairbank's group, and was abandoned when reliable fluorescence of  $\text{Ba}^+$  in LXe could not be observed [ref Kendy's thesis, after kamland in list].

Three barium tagging techniques continue to be explored. One of these is to grab the daughter on a surface, brought to the decay site by a probe, and then move it to a location where it can be desorbed from that surface by an infrared laser, and subsequently resonantly ionized by two lasers in order to detect it by time-of-flight spectroscopy. The apparatus for the study of this method is described in [17], along with some initial results. Another concept is to also grab the daughter on the surface of a probe, but then to thermally desorb it for subsequent trapping and detection [ref upcoming Liang paper, after Twelker].

The other remaining technique for barium tagging, now the concentration of Fairbank's group and the subject of this thesis, is called tagging in SXe. Here, a cold probe would go to the site of the candidate  $0\nu\beta\beta$  event in order to trap the  $\text{Ba}/\text{Ba}^+$  in a small amount of SXe,

and then observe it by its laser-induced fluorescence in the SXe, a technique called matrix isolation spectroscopy.

A concept for a probe is shown in Fig. [ref fig cryoprobe]. The SXe forms on a sapphire window or plug. Sapphire is a good candidate for a substrate; it has good thermal conductivity at low temperature, and is extremely transparent. An excitation laser would be brought into the probe through a fiber, and aimed through the sapphire to excite the Ba/Ba<sup>+</sup> in the SXe. A return fiber could collect the laser reflections to measure the SXe thickness via interference fringes. The Ba/Ba<sup>+</sup> fluorescence would then be collected by a lens/filter system and imaged onto a CCD. Additional components, not shown, would be required for cooling the sapphire, either by liquid He or by a Joule-Thompson nozzle, as well as a thermometer.

Whether the daughter Ba will neutralize or remain ionized is not yet known. It is however expected that a Ba<sup>++</sup> ion will neutralize once to Ba<sup>+</sup> in lXe, since the lXe conduction band gap is slightly less than the ionization potential for Ba<sup>+</sup> [10]. Further neutralization could also occur. A study of neutralization of alpha decay daughters in EXO-200 has observed a fairly equal balance of neutralized daughters vs. ionized daughters [ref alphaion]. The feasibility of detecting single Ba ions as well as atoms is still of interest.



FIGURE 1.12. Concept for a cryoprobe containing excitation/collection optics.

## CHAPTER 2

### THEORY

#### 2.1. BA/BA<sup>+</sup> SPECTROSCOPY IN VACUUM

The lowest-lying energy levels in vacuum for Ba and Ba<sup>+</sup> are shown in Fig. 2.1. For Ba, the main transition is between the ground  $6s^2 \ ^1S_0$  to the excited  $6s6p \ ^1P_1$  state. Spin(?)- suppressed transitions between the P state and three metastable D states results in a decay in to a D state after about 350 excitations. For Ba<sup>+</sup>, two strong transitions exist between the ground  $6s \ ^2S_{1/2}$  and the  $6p \ ^2P_{1/2}$  and  $6p \ ^2P_{3/2}$  excited states. Transitions to the two metastable D states are higher than for the atom, resulting in a decay into a D state after about 4 excitations.

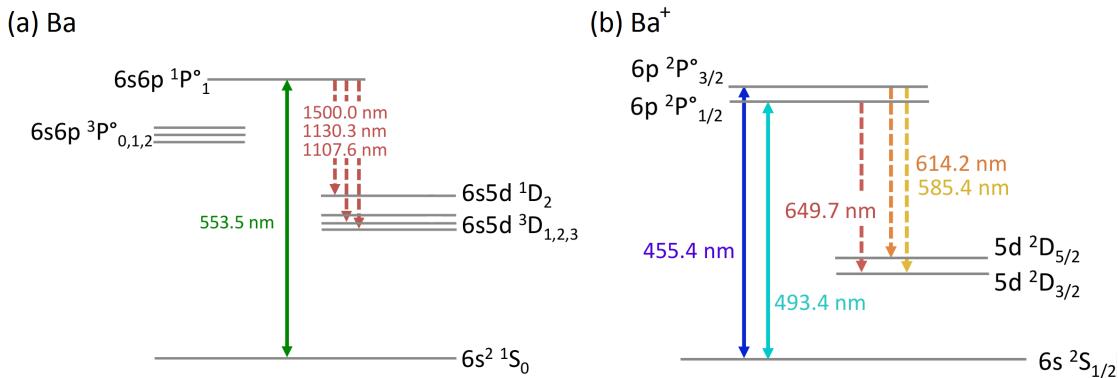


FIGURE 2.1. Energy level diagrams for (a) Ba and (b) Ba<sup>+</sup>

These energy levels and their transition rates are well known, and are documented in the NIST Atomic Spectra Database. Single atom/ion detection by spectroscopy generally requires, in addition to the main excitation laser, lasers to provide transitions out of the metastable D states once the atom/ion decays into one of them. For the atom, this requires three additional infrared lasers. Single atom trapping/detection [do they actually have

**single-atom in that paper? ... also they did it with a red laser too]** in a magneto-optical trap (MOT) is achieved in [20]. Single Ba<sup>+</sup> observation requires only two lasers if the <sup>2</sup>P<sub>1/2</sub> excited state is used. This is demonstrated in [ref something that does this ... is there something? How about the Carleton group? I think MOE has a reference to it maybe – yeah, its [19]].

## 2.2. MATRIX ISOLATION SPECTROSCOPY

In the solid matrix, the Ba/Ba<sup>+</sup> spectroscopy is affected by its interaction with neighboring Xe atoms.

*Kind of a main idea is the very different interaction between M and G for the ground vs. excited state. This leads to Franck-Condon rule and to Jahn-Teller. Shon doesn't really say that, I don't think. May just say that it's a van der waal force, w/o writing the force terms, though Crepin doesn't even seem to mention that as important.*

*May be good matrix isolation theory references in Ba Spec... yeah, 1 and 2 i think ... shon's 55, 57, 58 are probably good too and maybe overlapping*

*Does the lead into matrix isolation in Crepin start with just the interaction w/ noble gas? You could have a section for that like Shon, or just have it within this section.*

The spectroscopy of a species trapped in an inert solid matrix is called matrix isolation spectroscopy, the concept of which was pioneered around 19?? by [that one guy] [ref ... maybe [1] of ba spec]. Though absorption and emission are significantly broadened and shifted, a species can retain properties of its vacuum counterpart, such as quantum numbers.

Studies of some matrix isolation systems have been made. The most thoroughly studied has been Na(?) in solid matrices of (Ar, Kr, Xe ?)... (Na? Mg?)...[refs]

We are of course interested in the spectroscopy of Ba and/or Ba<sup>+</sup> in SXe matrices, and this particular system has not been studied until recently. The first report of the spectroscopy of neutral Ba in SXe, along with candidate fluorescence peaks for Ba<sup>+</sup> in SXe, was published by our group in [ref ba spec], and conversations following this publication with ???'s group in ??? have confirmed our basic observations of the absorption spectrum of Ba in SXe.

From here forward I will refer to the host species as Xe and the guest as Ba/Ba<sup>+</sup>, specific to our system. The leading interaction between [put this here if the van der waals is only necessarily the force for Ba in particular] the Ba/Ba<sup>+</sup> and a neighboring Xe atom is a (the) Van Der Waals (sp?) force, an induced dipole-dipole interaction. Xe is the most polarizable (sp?) of the noble gases.

The forms for this force(?) is shown in Eqn. [ref van der ba] for Ba and Eqn. [ref van der ba+] for Ba<sup>+</sup>. Talk about it.

This binding between the Ba/Ba<sup>+</sup> and its surrounding Xe atoms results in vibrational modes, and the Franck Condon (sp?) principle applies. Fig. [ref fig franckcondon] helps illustrate this effect. In a cold matrix, the system will be in the ground vibrational state before excitation. The distribution of the wavefunction, even for the ground state alone, overlaps in space in general with more than one of the excited state vibrational modes, resulting in a broadening in energy of the absorption.

Rapid decay occurs to the lowest vibrational mode in the excited state before electronic decay can occur [ref] **[is this just for Ba?]**. Then a similar broadening in emission energy occurs as several overlapping vibrational mode wavefunctions exist in the ground electronic state, and a redshift is also observed as some energy was dissipated into phonons in the crystal in vibrational mode decays. **(How can you get a blueshift?)**

Shifts and broadening in absorption and emission depend in general on the distribution of Xe atoms around the Ba/Ba<sup>+</sup>. The fcc crystal structure of the Xe restricts these environments to discrete number of so-called matrix sites, defined by (a) whether the Ba/Ba<sup>+</sup> is [intersituational or whatever – is that still a thing?], and (b) the number of vacancies in the Xe matrix surrounding the Ba/Ba<sup>+</sup>. Experimental observation of emission peaks from different matrix sites of [Na] in solid noble gases is reported in [ref(s)], with theoretical calculations in [ref(s)] attributing observed peaks to specific vacancy distributions defining the matrix sites (maybe this has been done?).

Energy level transition probabilities can also be affected in a matrix. Distortions in electron wavefunction shapes by asymmetric matrix sites can affect radiative transitions by altering parity [ref]. If electronic potential energy curves cross each other, nonradiative transitions can also become allowed for otherwise forbidden transitions [ref]. (Is a phonon emission called a nonradiative transition / does this actually happen?)

*Jahn-Teller?*

The detectability of a single atom/ion can depend greatly on altered transition rates. For example, in the Ba atom, matrix-allowed decay of the <sup>1</sup>P<sub>1</sub> to the <sup>3</sup>P states could be much stronger than the main transition back to ground, which would suppress fluorescence and make single-atom detection impossible. But the matrix could also help by strengthening decays from the D states back to ground, eliminating any need for re-pump lasers to keep the transition cycle going.

2.2.1. 6-LEVEL SYSTEM. Neutral Ba in vacuum is only a 5-level system. However, solving a 6-level system is helpful in modeling matrix-isolated Ba where interactions could allow a transition into a <sup>3</sup>P state.

### 2.3. SHON HAD A FLU EFF SECTION

## CHAPTER 3

# EXPERIMENTAL

*Probably need to mention CCD digitaztion pedestal and noise, and dark counts which are zero when cold – did you do that?.*

This chapter describes the apparatus at Colorado State University which has been used for all described studies of Ba/Ba<sup>+</sup> fluorescence in SXe after deposition in vacuum. Our main barium source, a Ba<sup>+</sup> ion beam, is first described, as well as a purely Ba neutral source. The co-deposit of Ba/Ba<sup>+</sup> with Xe gas onto a cold sapphire window, subsequent laser excitation, and finally the collection optics for the fluorescence are described.

### 3.1. ION BEAM

The full ion beam is shown in Fig. 3.1. This is a clean source of Ba<sup>+</sup> which can do a very wide range of deposit sizes, from billions of ions in a focused laser region all the way down to the single-ion level, and even deposit sparsely enough to scan a focused laser over spatially separated ions.

**3.1.1. ION SOURCE.** Ba<sup>+</sup> ions are produced in a Colutron ion gun system [21]. The source is shown in Fig. 3.2. A solid barium charge is placed into the hollowed end of a stainless steel rod, which is then inserted into the discharge chamber, near the hot filament. The barium vaporizes, and escapes the hollowed rod around a loosely threaded set screw. The source is designed to produce a discharge between the anode plate and the filament cathode, through an argon buffer gas leaked into the source chamber. This controlled discharge would then also ionize atoms from the solid charge to produce the desired ion beam, and Ar ions would be filtered out. However, to avoid contamination of the SXe matrix with residual

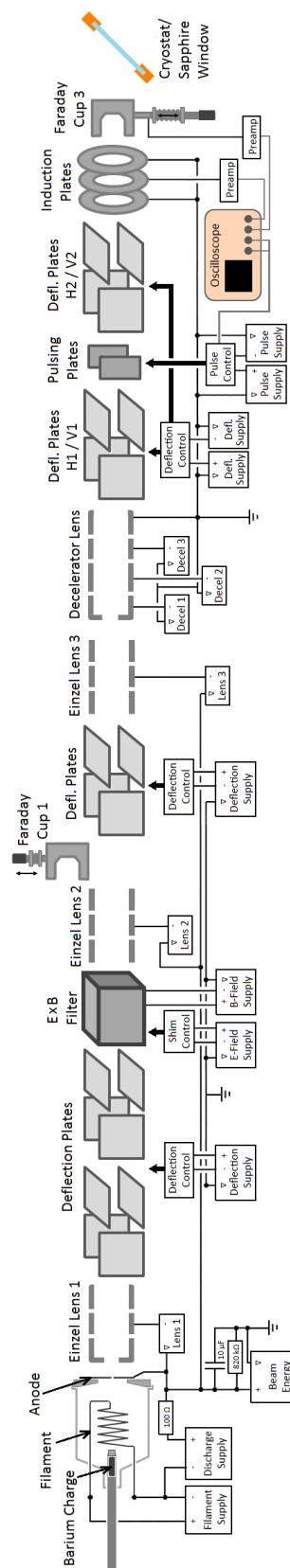


FIGURE 3.1.  $\text{Ba}^+$  ion beam.



FIGURE 3.2.  $\text{Ba}^+/\text{Ar}^+$  ion source.

Ar gas, the buffer gas is not used in this work. We are still able to maintain a discharge between the filament and anode circuits. The longevity of ion current from a single charge (at least several 10s of hours) suggests that Ba is coating the inner walls of the chamber and is depleted slowly. This is supported by the observation of white oxidation of the inner source parts after a few minutes of exposure to air when opening the system. The discharge produces a plasma, containing barium ions, which escapes the chamber through a small hole in the anode, where it enters the acceleration region. The acceleration potential is 2 kV, between the ion source anode and an aperture, which is the first element of Einzel lens 1. This lens approximately collimates the ion beam for passage through the  $E \times B$  velocity filter.

**3.1.2.  $E \times B$  VELOCITY FILTER.** The  $E \times B$  velocity filter selects  $\text{Ba}^+$  by creating perpendicular electric and magnetic fields, which produce opposing forces on charged particles moving straight through the filter. The opposing forces will be equal for ions with velocity

$v = \frac{E}{B}$ . Since ion velocity is determined by mass ( $m$ ), charge ( $q$ ) and beam potential ( $V$ ), the filter selects ions satisfying Eq. 4:

$$(4) \quad \frac{m}{q} = \frac{2VB^2}{E^2}$$

where  $B$  and  $E$  are the magnetic and electric fields, respectively. Those fields are chosen such that the forces are equal for  $\text{Ba}^+$ . Other ions will be deflected.

The  $E \times B$  filter is shown in Fig. 3.3. Electromagnets provide the vertical magnetic field. Electrode plates and field-shaping guard rings provide the horizontal electric field. The guard rings prevent a lensing and astigmatism effect from fringe fields of the plates [21].

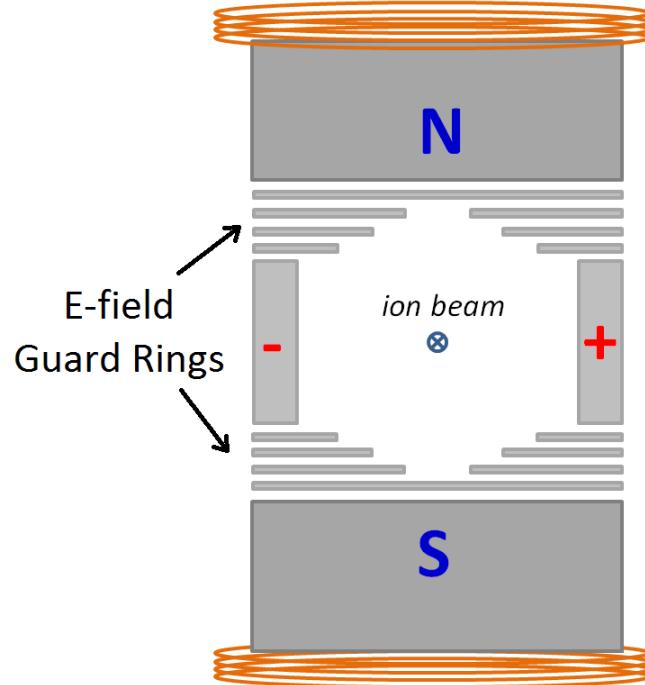


FIGURE 3.3. Colutron  $E \times B$  ion velocity filter.

3.1.3. OTHER BEAM COMPONENTS. The first three sets of deflection plates can be used for beam diagnostics, and are set to 0 V during normal operation. The deflection plates just

before the pulsing plates, H1 and V1, are set to constant values of 50 V and 0 V, respectively, which have been selected such that the beam, in both pulsing and continuous modes, can be deposited at the sapphire window for reasonable settings on the final deflection plates, H2 and V2. As described in 3.1.5, different settings in H2/V2 are required for peak ion current in Faraday cup 3 vs. peak deposit at the window.

Einzel lens 2 focuses the beam to pass through the aperture in the first element of the decelerator lens. Einzel lens 3 is not used in this setup. The decelerator lens can be used to vary  $\text{Ba}^+$  deposit energy, but in this work it acts as an Einzel lens with only the second element at voltage, and it focuses the beam at Faraday cup 3 (there is no Faraday cup 2 in this setup). Faraday cup 3 measures the ion current during experiments, and is retracted when deposits are being made. Its use in deposit calibration is described in 3.1.5. Faraday cup 1 is used for beam diagnostics, and is usually retracted on its bellows.

3.1.4. ION BEAM PULSING. When running in pulsing mode, the pulsing plates are first set to 200 V and -200 V to deflect the beam, and are pulsed to 0 V for 1  $\mu\text{s}$  for each pulse. The pulsing circuit is shown in Fig. 3.4. Square waves, triggered by LabVIEW at 500 Hz, enter the circuit at (a). The transformed pulse triggers the MOSFET switch, which closes the circuit for the period of the pulse.

The induction plates observe pulses during a deposit. Pulses just prior to a deposit can be observed by cup 3 (as well as the induction plates) for a local measurement of ion current in the pulses. eV Products pre-amplifiers convert the ion current to voltage signals which are read out with a digital oscilloscope. An example of an oscilloscope readout of 16 averaged pulses is shown in Fig. 3.5(left). The pre-amp circuit is solved to retrieve the original current, given by Eqn. 5:

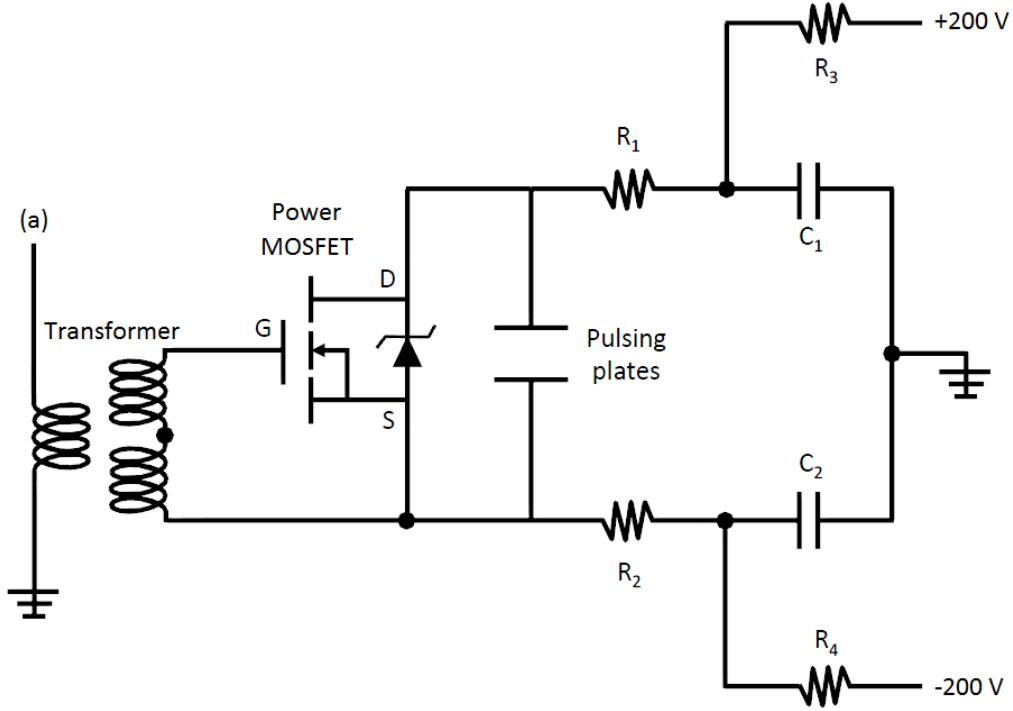


FIGURE 3.4. Pulsing circuit.  $R_1 = R_2 = 470 \Omega$ ,  $R_3 = R_4 = 20 \text{ k}\Omega$ ,  $C_1 = C_2 = 680 \text{ nF}$ . [22]

$$(5) \quad I = \frac{-(V_{out} + R_1 C \frac{dV_{out}}{dt})}{R_1 M}.$$

$R_1 C$  and  $R_1 M$  are determined for each pre-amp. The time constant  $R_1 C$  is determined by fitting an exponential to the decay of a signal, and  $R_1 M$  is a constant which is then determined by fitting this shaped signal to a known input square pulse. Final shaped currents for signals from the induction plates and cup 3 are shown in Fig. 3.5(right).

Pulsing data also provides confirmation that the beam is composed of  $\text{Ba}^+$ . The time between the center of the pulsing plate voltage overlap and the center of the pulse measured by the Faraday cup, along with a measurement of the distance traveled, provides a velocity measurement of the ions. This distance was measured to be  $31.5 \pm 0.5 \text{ cm}$ , and time-of-flight



FIGURE 3.5. Raw (left) and shaped (right) pulse signals from induction plates and cup 3. The raw induction signal appears small because it is a less sensitive pre-amp (accounted for in shaping).

data, e.g. Fig. 3.6, give  $39.8 \pm 3.4$  amu for  $\text{Ar}^+$  and  $136.8 \pm 6.3$  amu for  $\text{Ba}^+$ , including an uncertainty on the time of flight of  $\pm 0.1 \mu\text{s}$ . This rules out any  $\text{BaO}^+$  contribution to the  $\text{Ba}^+$  ion beam.

**3.1.5. CALIBRATION OF ION DEPOSIT.** To calibrate the signal at cup 3 to an ion density at the sapphire window, another Faraday cup (cup w) is attached to the cold finger in place of the sapphire window. Firstly, the ratio in  $\frac{fC}{pulse}$  between cup 3 and cup w is measured ( $\equiv f$ ). Then, knowing the radius of cup w lets one determine the ion density per pulse at the sapphire window:

$$(6) \quad \frac{ions}{pulse \times m^2} = \frac{Cf}{Ae}$$

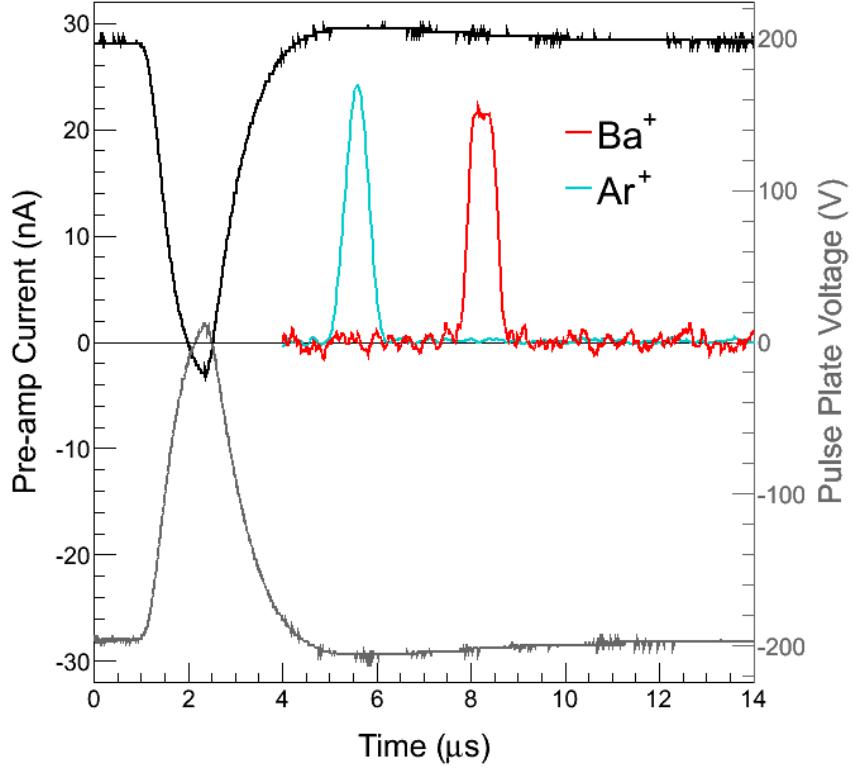


FIGURE 3.6. Arrival time of pulses at cup 3 vs. time of pulsing plate signal (black (+) and gray (-)) for  $\text{Ar}^+$  and  $\text{Ba}^+$ .

where  $C$  is  $\frac{fC}{pulse}$  at cup 3,  $A$  is the area of cup w, and  $e$  is the elementary charge. The required settings of the final deflection plates H2 and V2 are also determined by cup w. These typically differ from the peak values for cup 3 by about 70 V in H2 and 60 V in V2, corresponding to about 4 mm in x and y position at cup w.

### 3.2. BA GETTER SOURCE

A  $\text{BaAl}_4$  getter is employed as a purely neutral Ba source which can be inserted on a bellows to emit toward the sapphire window, shown in Fig. 3.7. Getters were used extensively in previous work [23] for measuring the absorption of Ba in SXe with large Ba deposits, and in identifying emission peaks. A getter is used briefly in this work in identifying the 619-nm fluorescence peak, as described in 4.3.4. Getters used in [23] were exothermic  $\text{BaAl}_4\text{-Ni}$



FIGURE 3.7. Apparatus around sapphire window, including Ba getter, and optics for excitation and collection.

flash getters. The getter used here is an endothermic type, designed for more controlled Ba emission.

### 3.3. SAMPLE DEPOSITION

The  $\text{Ba}^+/\text{Ba}$  is co-deposited with ultra-pure Xe gas onto a cold sapphire window. Sapphire has good thermal conductivity at low temperature and good optical transparency in the visible. The window is held to a coldfinger and is tilted at  $45^\circ$  to allow access of the ion beam as well as the excitation laser and collection optics. To begin a deposit, Xe gas



FIGURE 3.8. Setup for measuring SXe growth rate by interference fringes.

is flowed toward the window via a leak valve. Cup 3 is then retracted to clear the path for  $\text{Ba}^+$  ions, which land in the SXe matrix as it grows. Cup 3 is then replaced, and the Xe leak stopped.

SXE matrix growth rate can be measured by interference fringes in a laser reflected against the front surface of the sapphire window, as shown in Fig. 3.8. Fringes for SXe deposition at 52 K and 11 K are shown in Fig. 3.9 for the leak rate used in this work. The refractive index of SXe has a negligible dependence on temperature between 50 and 30 K [24], so these can be compared directly. A somewhat lower rate is observed at 52 K, at 31 nm/s, vs. 37 nm/s at 11 K. To evaporate a sample, the window is heated to 100 K. Fringes appear during this process as well. The full set of fringes for a deposit at 52 K and its evaporation when heated is shown in Fig. 3.10, along with the window temperature. This shows that the SXe evaporates between 73 K and 78 K. Notice that the same number of fringes appear in the deposit and the evaporation.

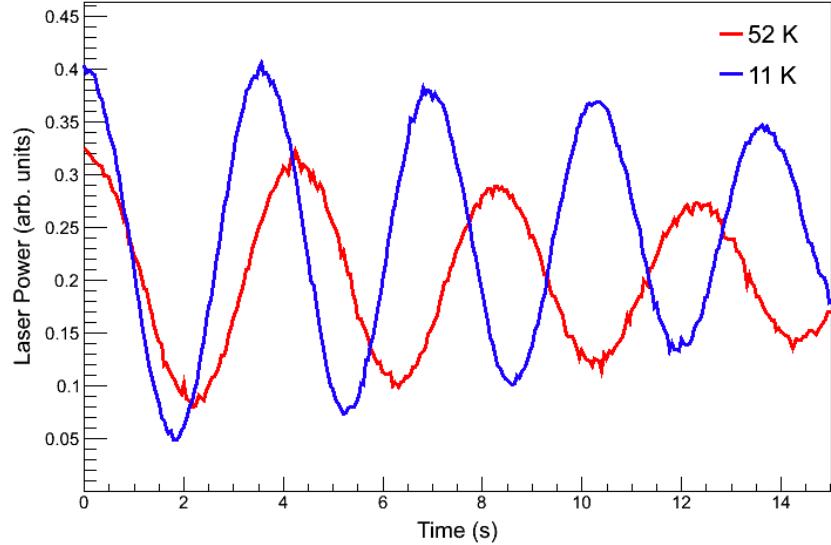


FIGURE 3.9. Interference fringes for the same Xe gas leak rate deposited on the sapphire window at 11 K and at 52 K.

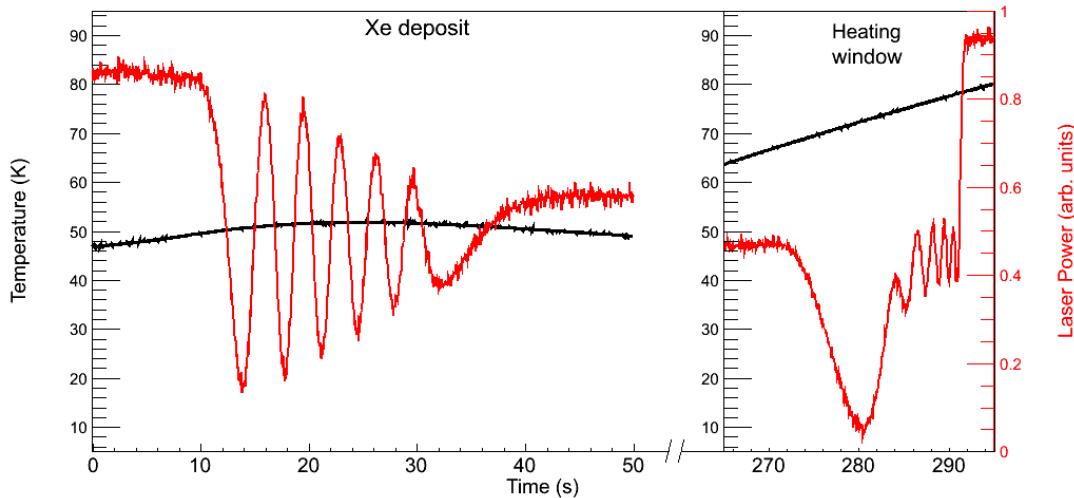


FIGURE 3.10. Interference fringes of a deposit at 52 K and of its subsequent evaporation when heating the sapphire window.

### 3.4. LASER EXCITATION

Green/yellow laser excitation is done with a Coherent 599 dye laser, pumped by the 514-nm line of a Lexel 3500 Ar ion laser. Rhodamine 110 (R110) dye tunes over 542 - 566 nm, and Rhodamine 6G (R6G) tunes over 567 - 590 nm. Another Coherent 599 contains Coumarin 480 (C480) dye for blue excitation, which is pumped by a Kr ion laser.

A 7.9-cm focal length aspherical lens focuses the laser into the cryostat from the side (Fig. 3.7). This lens reduces spherical aberration in the laser focus so that it can reach the diffraction limit. However, to reach the diffraction limit in the SXe, astigmatism introduced by the tilted sapphire window must be corrected. To do this, compensating astigmatism is introduced by a fused silica optical flat of 1 cm thickness, placed after the lens, tilted in the y-plane, whereas the sapphire window is tilted in the x-plane. The proper angle for the compensator is determined to be about  $10^\circ$  from normal by a ray matrix equation for a sapphire window of thickness 0.5 mm [25]. The effect of the SXe layer is negligible since its thickness is less than a wavelength. With the compensator, overlapping minimum spot sizes of  $2.06 \mu\text{m}$  and  $2.66 \mu\text{m}$  are calculated for x and y, respectively, corresponding to an effective minimum spot size ( $w_0$ ) of  $2.36 \mu\text{m}$  in the SXe. *Show astig. corr. data w/ 2D gaus fits.*

### 3.5. COLLECTION OPTICS

Fluorescence is collected above the cryostat (Fig. 3.7). A 50 mm Nikon camera lens collimates the light, and a fluorescence filter sits on top of it. A band-pass filter is used for imaging, and a Raman filter is used for spectroscopy. The fluorescence then reflects off two steering mirrors, and is imaged by a 200 mm Nikon camera lens onto a Roper Scientific liquid-nitrogen-cooled CCD, with a magnification of 4.

The CCD has a removable Princeton Instruments imaging spectrometer. With this attached, the 200 mm camera lens focuses the light onto an inlet slit, which is then imaged by the spectrometer onto the CCD after reflecting off a diffraction grating. The 0-order reflection of the grating provides an image for alignment, and the grating can be tilted to distribute the 1<sup>st</sup>-order reflection across the horizontal CCD pixels for doing spectroscopy.



FIGURE 3.11. Put pic of new ones? Otherwise w/ asphere at least.

When a 1" diameter filter is used, the system has a collection efficiency ( $\epsilon_c$ ) of about 1.5% without the spectrometer, and about half of that with the spectrometer due to a grating efficiency of 50%. The spectrometer also limits the system to an f-number of 4.

To limit laser exposure to only the time of CCD exposure, a laser shutter is linked to the camera exposure with a LabVIEW program. This program also records laser power via a calibrated pickoff, as well as the temperature of the coldfinger near the sapphire window during observation.

### 3.6. LASER SCANNING

In order to obtain images of separated single atoms, the laser focusing lens is attached to motorized Newport translation stages which scan the laser position by scanning the lens in x and y, shown in Fig. 3.11. These stages sit atop a manual z-translation stage for laser focusing. An extension of the aforementioned LabVIEW program coordinates movement of these stages such that x or y position is stepped in between CCD frames, and each frame then corresponds to a position in a laser scan grid.

## CHAPTER 4

# RESULTS

Where did you say to write how defocused alignment is done (i.e., find focus, then defocus and center fluorescence) ... IT'S IN BLEACHING AS "SHOULD SAY HERE". I think it should be said before discussion (or w/) of its first usage, which may be excit spec rather than bleaching.

Studies of several emission peaks of neutral Ba in SXe are discussed in 4.1. Temperature and annealing dependence are discussed in 4.1.1, and bleaching of these peaks is discussed in detail in 4.1.2. Imaging of Ba fluorescence in a focused laser region is discussed in 4.3, with the ultimate achievement of imaging at the single atom level using the 619-nm fluorescence peak. Candidate fluorescence peaks of Ba<sup>+</sup> in SXe are reported in 4.4.

### 4.1. FLUORESCENCE OF BA IN SXE

Deposits of Ba in SXe absorb primarily between 540 nm and 570 nm. An absorption spectrum, obtained by observing absorption of white light by a large Ba deposit at 11 K, is shown in Fig. 4.1, along with an example emission spectrum with excitation at 557 nm. Significant broadening, as well as a 4-nm redshift of the central peak, occur relative to the vacuum  $6s^2 \ ^1S_0 \rightarrow 6s6p \ ^1P_1$  absorption value of 553.5 nm. Initial discovery of this absorption and emission was done with the purely neutral Ba getter source. Observation of the same spectra from Ba<sup>+</sup> ion beam deposits demonstrates some neutralization of the ions. The fraction of ions neutralized is not determined. [18, 22, 23]

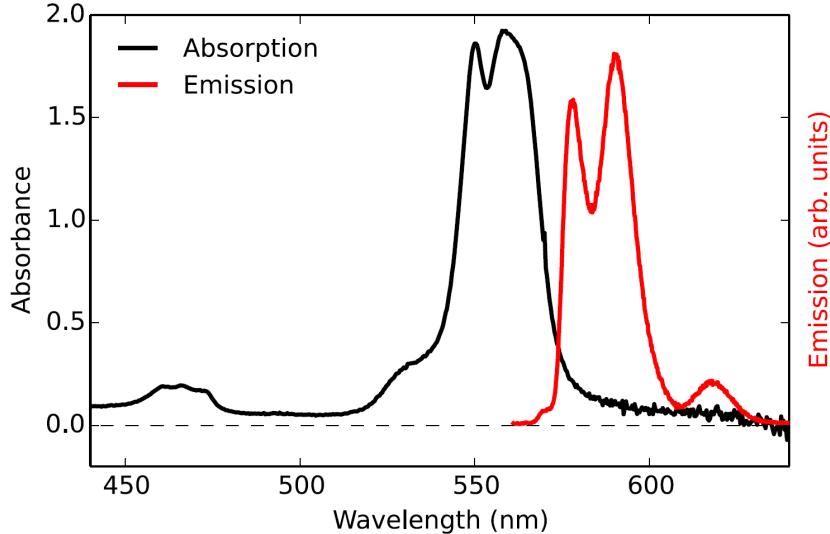


FIGURE 4.1. Absorption and emission spectra of neutral Ba in SXe. [18]

The several red-shifted emission peaks observed are attributed to Ba atoms occupying different matrix sites in the SXe. Deposition at temperatures higher than 11 K and exploration of more excitation wavelengths have led to discovery of a few emission peaks beyond the 591- and 577-nm peaks reported in [22] and [23]. Emission spectra for selected excitation wavelengths are shown in Fig. 4.2(a), for a deposit made at 44 K and observed at 11 K. These selections are part of a full excitation spectrum, shown in Fig. 4.2(c). An excitation spectrum is produced by scanning the dye laser and measuring the magnitude of each fluorescence peak vs. excitation wavelength. For each 1-s CCD exposure, a sum of asymmetric Gaussian functions and standard Gaussian functions is fit to the spectrum after pedestal subtraction. The asymmetric Gaussian fit function used, which fits the 577- and 591-nm peaks somewhat better than standard Gaussians, is  $A(1+\text{erf}(\frac{x-a}{\sigma_1})(1-\text{erf}(\frac{x-a}{\sigma_2}))$ , where  $A$  is the free amplitude parameter. The peak position  $a$  is fixed for each peak, as are the left and right smearing widths  $\sigma_1$  and  $\sigma_2$ . The function  $\text{erf}()$  is a Gaussian error function. The asymmetric Gaussian function is not perfect for all frames, as the shape of

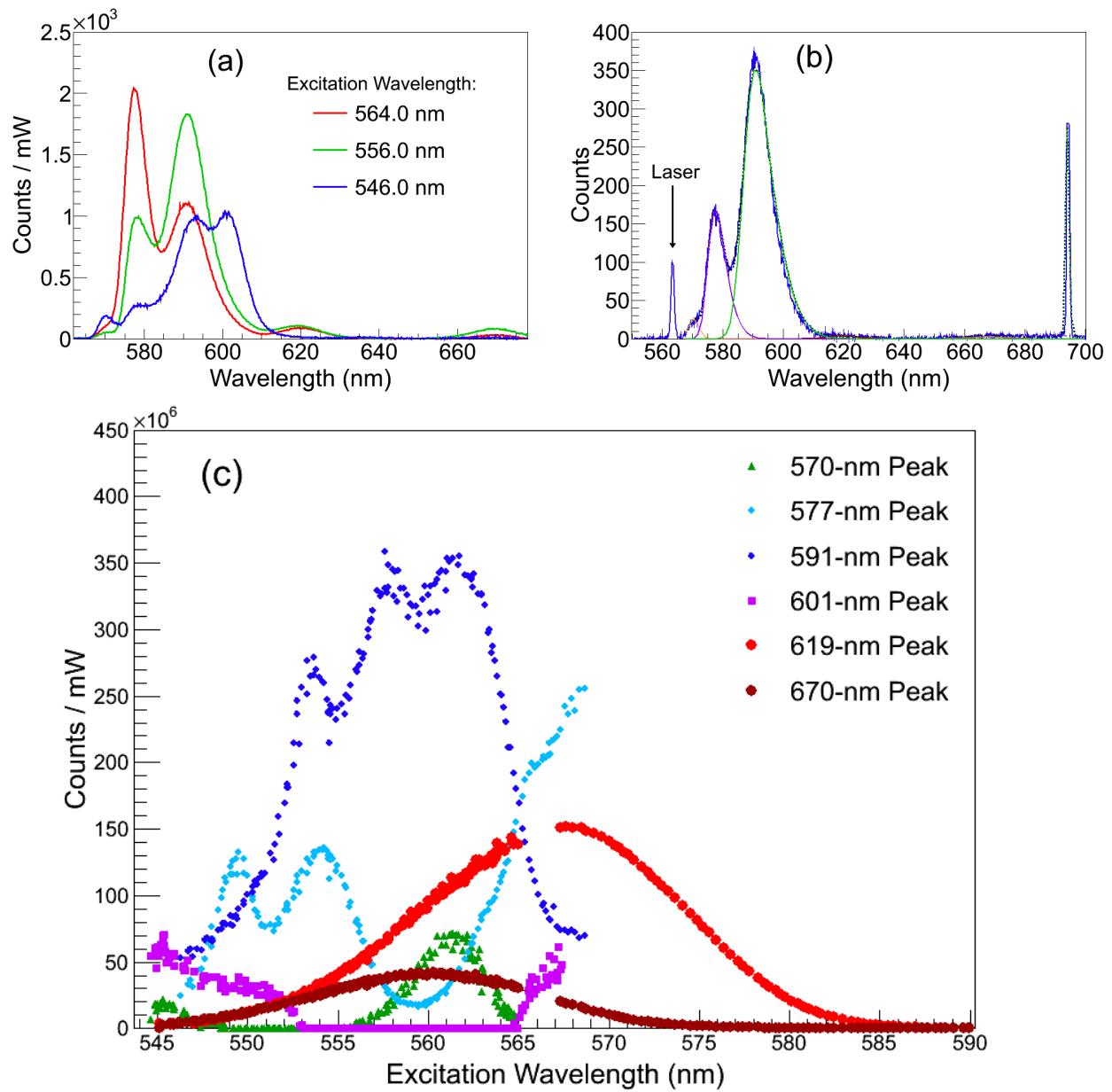


FIGURE 4.2. *Put legend in (b)* Background-subtracted fluorescence spectra for a few different excitation wavelengths (a), an example fit to a spectrum (b), and full excitation spectra for all observed Ba fluorescence peaks (c). Magnitudes in (c) have been scaled for visibility on the same plot (relative magnitudes are arbitrary as they are affected by relative site populations and fluorescence efficiencies). The discontinuity around 566 nm for the 619- and 670-nm peaks is the boundary between usage of different laser dyes. R6G dye is used for higher wavelengths and R110 for lower wavelengths and for all other curves. Curves for the R6G segments (619- and 670-nm peaks) require special scaling to line up with their respective R110 segments, and this scaling was different between the 619- and 670-nm peaks, likely due to different relative populations of those sites on the different deposits.

the emission can depend on the excitation wavelength. An example frame with fits is shown in Fig. 4.2(b). The full fit is the dotted black line, and each peak's contribution is in color. Curves in (a) have about  $100\times$  the laser power used in the excitation spectrum (e.g. (b)) where low intensity is desired to avoid bleaching during the scan. Rather than attempting frame-by-frame background subtractions, additional Gaussians are fit to the broad and sharp background fluorescence. These backgrounds and their excitation spectra are discussed in 4.2. A 566 nm Raman filter was used to attenuate the majority of the laser scatter, however the small amount of scatter passed by the filter was used to determine the frame's excitation wavelength. Each peak's fit contribution was then integrated and scaled by the frame's laser power, as the power output is not constant through the dye range.

Wavelength calibration was done using three lasers whose wavelengths are first measured with a Burleigh Wavemeter: a red diode laser at 656.99 nm, a doubled Nd:YAG laser at 532.23 nm, and the C480 blue dye laser typically around 475 nm (at 488.91 nm for the data in Fig. 4.2). These lasers were directed at the same position on the sapphire window, and their scatter is imaged along the same path as the Ba fluorescence. The WinSpec software applies the diffraction grating equation to calibrate each CCD pixel to a wavelength.

**4.1.1. ANNEALING/TEMPERATURE DEPENDENCE.** Matrix site occupancies for Ba atoms can depend on annealing history, and similarly on the temperature at which a deposit is made. Spectra under different temperature conditions are shown in Fig. 4.3. Peak shapes look similar for deposits made at 40-55 K as those made at 11 K and then annealed to 40-55 K, both being observed at 11 K, though relative amplitudes are not the same, likely due to matrix site populations. Without annealing, 11-K deposits have a different peak shape around 590 nm, with a broader peak centered around 596 nm. Another possibility is that

the broader shape is due to a higher population of the 601-nm peak, and it is not resolved from the 591-nm peak. The experiment in Fig. 4.3 did not have optimal laser intensity for observing behavior of the 619-nm peak; a study of deposit conditions for this peak is shown in Fig. 4.4. In that experiment, the 619-nm fluorescence was observed in a focused laser beam at 570 nm, in an image through the 620-nm band-pass filter. About  $3\times$  more 619-nm fluorescence was observed in the deposit at 52 K vs. 11 K for the same leak rate (the respective 31 nm/s and 37 nm/s are from the same Xe leak rate, resulting in different SXe deposition rates according to Fig. 3.9). Also in this figure is a deposit at 11 K with the lower leak rate resulting in about 5 nm/s SXe deposition, which produced nearly  $3\times$  lower signal than the leak rate resulting in 37 nm/s SXe deposition. (*re-?)cite the 1:1e3 or whatever paper?* Xe leak rates above 37 nm/s result in rapid frosting of the SXe matrix, which causes blurring of the image and high laser scatter. These tests guided the standard of depositing with Xe leak rate resulting in 31 nm/s SXe deposition at  $50\pm5$  K when observing the 577-, 591-, and/or 619-nm peaks. The different bleaching behavior in a low Xe leak rate deposit is discussed in 4.1.2.

Fluorescence spectra with 564 nm excitation through several annealing cycles for a deposit made at 11 K are shown in Fig. 4.5. At this wavelength, all observed Ba peaks are prominent except the 570-nm peak. Similar to excitation spectra, individual fluorescence peak amplitudes can be plotted vs. temperature by fitting the full spectrum in each frame. Rather than asymmetric Gaussians, this analysis used standard Gaussian and Lorentzian functions. Lorentzians fit the 619-nm and 670-nm peaks well. Each individual peak's amplitude is plotted vs. temperature in Fig. 4.6(b-f), with an example of a fit spectrum in Fig. 4.6(a). In this interpretation, the 601-nm peak along with the 591-nm peak fit the broader

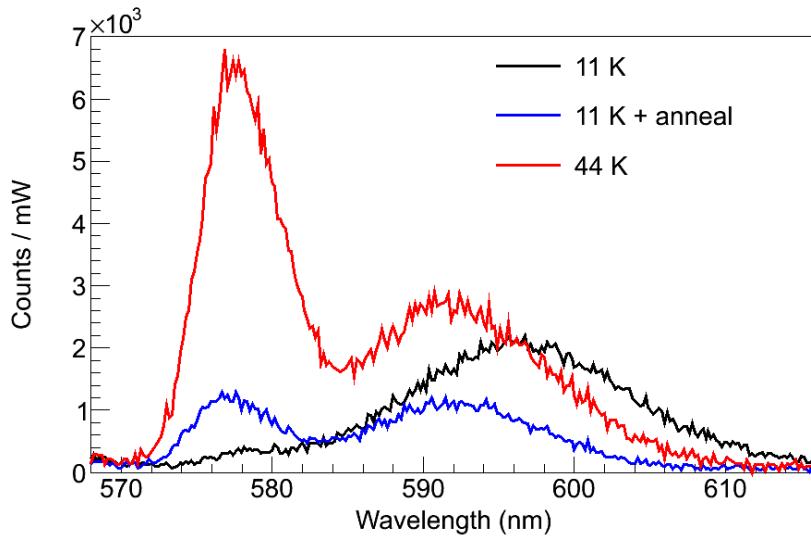


FIGURE 4.3. Spectra of peaks around 590 nm of a  $\text{Ba}^+$  deposit made at 11 K before and after annealing to 39.4 K, and one made at 44 K. All observations are at 11 K. Both  $\text{Ba}^+$  deposits are 15 s, however the 44 K deposit is scaled slightly to account for different ion current. Laser power was about 0.1 mW with an unfocused beam waist of  $w = 7.056$  mm, at 566 nm wavelength.

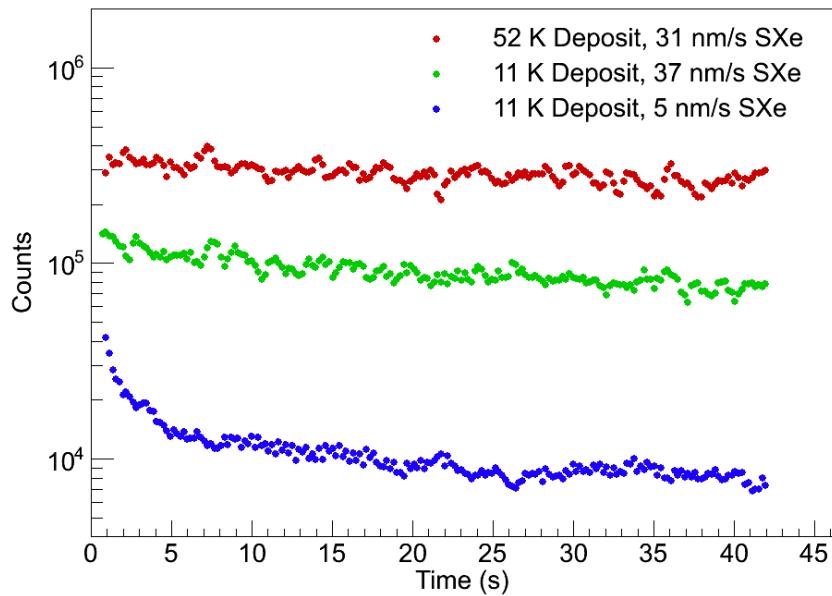


FIGURE 4.4. Summed 619-nm fluorescence counts over time from imaging focused 570-nm laser region, for  $\text{Ba}^+$  deposits at different temperature and Xe leak conditions. Frame times are 0.1 s, with time in between frames of about 0.11 s.

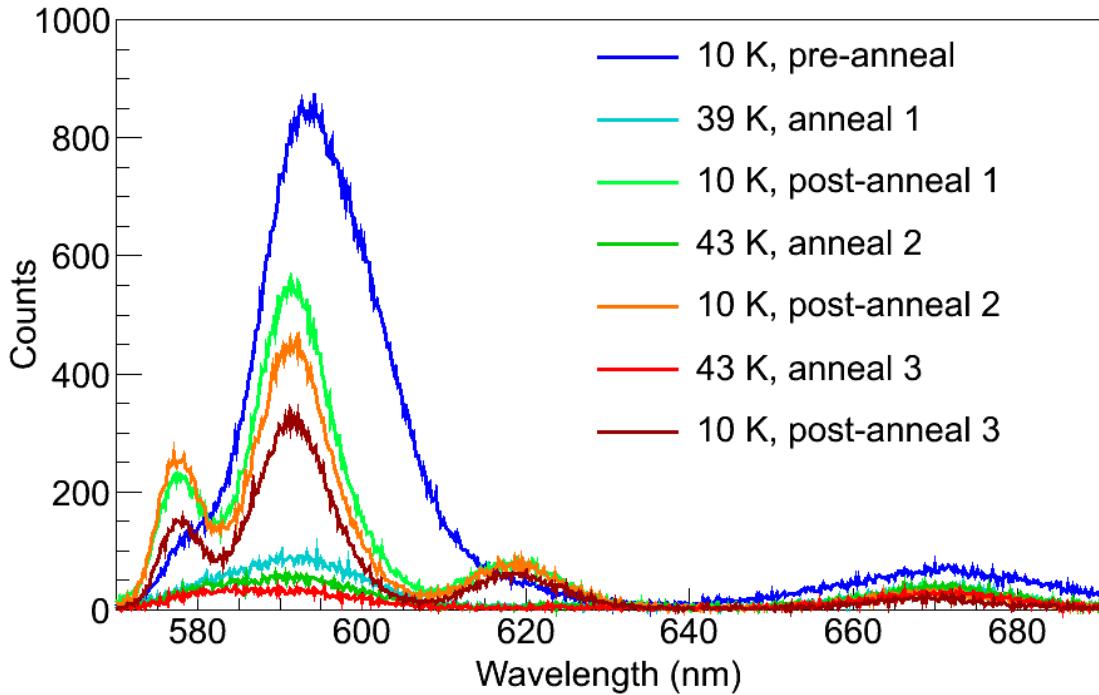


FIGURE 4.5. Spectra of a large  $\text{Ba}^+$  deposit through several annealing cycles. Initial deposit was at 11 K. Laser power was about 0.2 mW with an unfocused beam waist of  $w = 7.056$  mm, at 564 nm wavelength. These are selections from the full data set shown as peak counts vs. temperature in Fig. 4.6. *This same data, but a different plot, is in BaSpec, as is some of the annealing plots – do I just cite the paper or what?*

peak in the initial 11 K deposit. The 601-nm peak shows nearly complete loss in the first anneal cycle. The 670-nm peak sees significant loss, with more loss with each succeeding anneal cycle, each of which reach higher temperatures. The 591-nm sees moderate loss with each cycle. The 577-nm and 619-nm peaks gain significantly with the first anneal, suggesting that they are due to more stable matrix sites. Both peaks remain about the same after the second cycle (small gain in 577-nm), and both see loss after the third cycle, which reaches the higher temperature of 48 K. *Either just say this could be due to diffusion (or something else?), or was there a specific temperature to reference from that paper?*

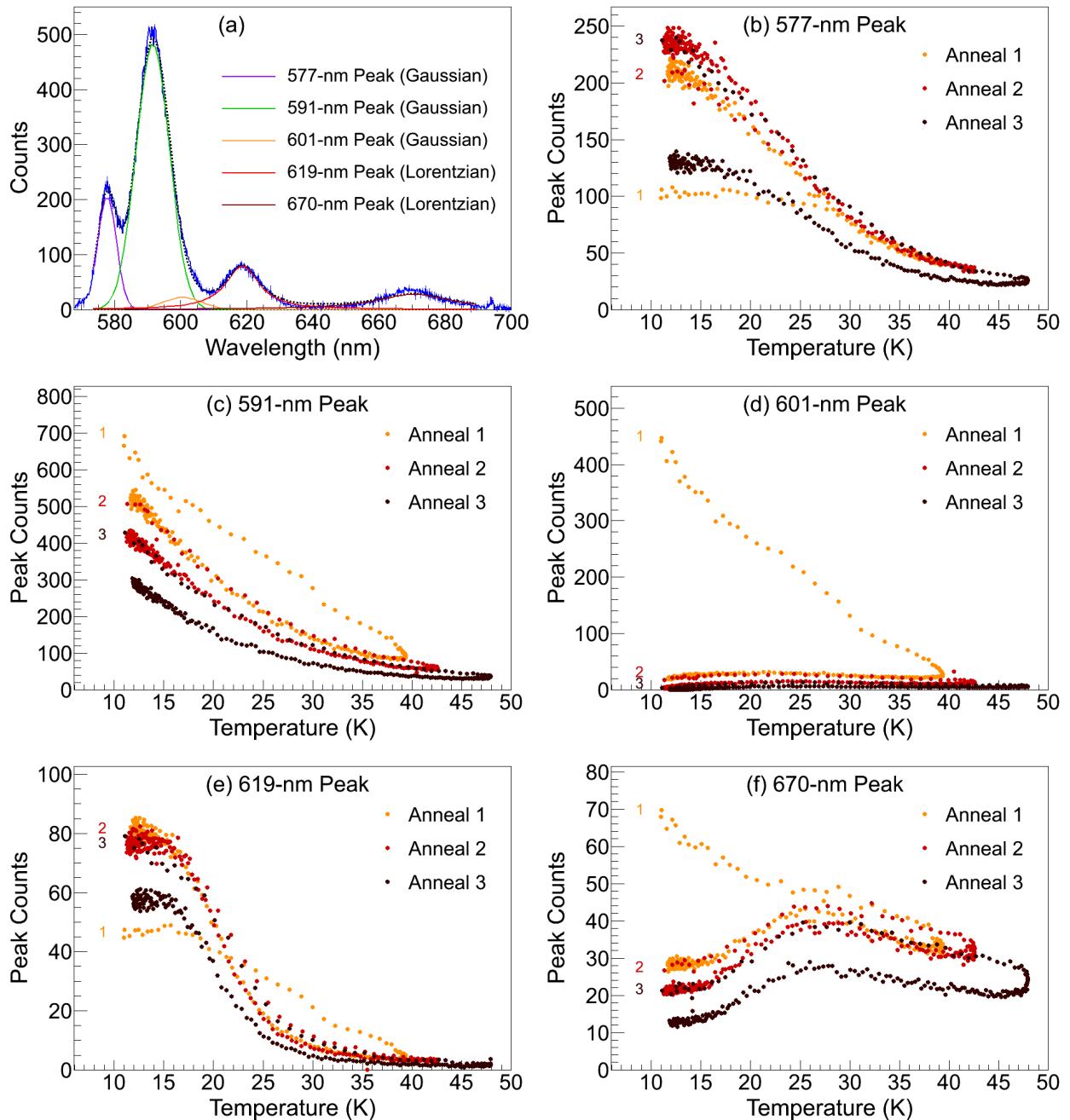


FIGURE 4.6. Example fit with standard Gaussians and Lorentzians (a) and fit peak counts for each fluorescence peak through three annealing cycles of the  $\text{Ba}^+$  deposit made at 11 K (b-f). “1, 2, and 3” mark the beginning of each anneal cycle. Laser power was about 0.2 mW with an unfocused beam waist of  $w = 7.056$  mm, at 564 nm wavelength.

Aside from matrix site changes, direct temperature dependence of fluorescence is observed in annealing cycles. The 577-, 591- and 619-nm peaks have their highest amplitude at 11 K.

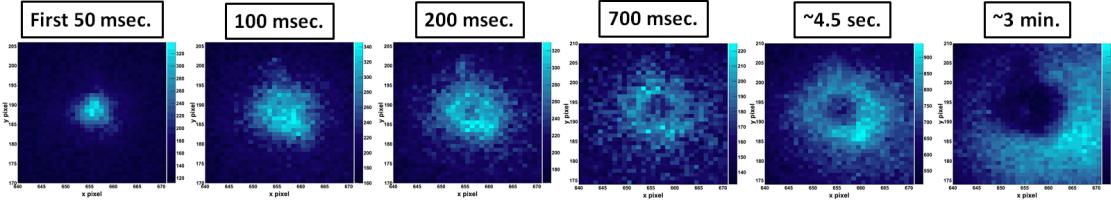


FIGURE 4.7

The 577-nm and 619-nm peaks appear to be leveled off at 11 K, while the 591-nm may benefit from even lower temperatures. The 670-nm peak has its highest amplitude at around 25 K (apparent after its major site alteration has occurred). This inverse relationship between fluorescence and temperature suggests that a probe in nEXO will need to be moved to an evacuated chamber in order to cool to 11 K or below for observation.

**4.1.2. BLEACHING.** *This section is referred to in annealing section as talking about different bleaching in leak 44 runs.*

Decay of fluorescence with laser exposure, or bleaching,  
*um, or not, ala other statements: Describe how you align here – observe focused image, find its focus, and defocus by an amount, then centering the broad Ba fluorescence for a known intensity region of x% ... do you want an explanation of the image in experimental? or maybe this is a good place to introduce it?*

(do a correction on p-meter sensitive area and on p-meter quantum efficiency, and also a spherical aberation correction for power, though that may not matter for the defocused studies.)

590 etc, model fit (see results intro paragraph)

Repump attempts – nothing found with cheap lasers.

619, with the changes in time and I

bleaching effect on excit spec (bleaching site change search)

## 4.2. BACKGROUNDS

*Says in excitspec sec that here you talk about BG excitspec, and you should say how it's used with 619's to choose 570 nm. ... what about the surface BG? says it's similar?*

One source of background in spectra and images is fluorescence from the sapphire window. A spectrum of this fluorescence with x nm excitation is shown in Fig. [fig sapphire spectrum](a). The strong, sharp peak at *(peaks around? May need to ask Bill about what peaks exist and at what temps a new idea is to just leave it vague, as "peaks" since you're sure there are at least 2, but vague is OK)... You had 2-peak fits in one, right? was that blue? cuz i thought it was only warm that you observed both ... ??* x nm are a well-known fluorescence of Cr<sup>3+</sup> impurities in the sapphire bulk. *Is only one of the lines (the one mentioned on wikipedia) strong at room temp? I think you have room temp spectra somewhere.* A weaker and much broader fluorescence is observed, also from the sapphire bulk, which reaches into the wavelength regions of the Ba fluorescence peaks. The excitation spectra for these fluorescence components are shown in Fig. [fig Cr](b), over the range of both R6G and R110 dyes, as well as the blue dye C480. The excitation spectrum for the **693(s?)** Cr<sup>3+</sup> peak(s) agrees with the literature [cite that one paper]. The identical excitation spectrum of the broad fluorescence, including the small peaks around x nm, demonstrate that it too is due to Cr<sup>3+</sup> in the sapphire. *check this – was it certainly from just the bulk and not from partly surface BG for those excit. spec.? – ACTUALLY is it possible that some of this is the surface BG? Yes, if it was a broad laser, unless the sapphire is really high in these, which maybe it is?.* Commercially available c-plane quality sapphire windows contain very low concentrations

of Cr<sup>3+</sup>. The highest purity windows we have found are from Meller Optics, with Cr<sup>3+</sup> concentrations of around 10 ppt estimated from our spectra.

Another background emission is observed from the surfaces of the window. Its broad fluorescence is shown in Fig. [fig 1st order surface BG] with a 610-nm Raman filter cutoff and 570 nm excitation, in a 1<sup>st</sup>-order image, where the x-axis is wavelength and the y-axis is y position in the image. The nature of this emission has not been determined, however a few features have been identified. One is that the emission increases as the window temperature is decreased, down to about 100 K where it is flat down to 11 K, shown in Fig. [fig BG temp dep] along with the temperature dependence of the Cr<sup>3+</sup> emission, which has the inverse relationship. Another feature of the surface background is that it bleaches with laser exposure, shown in Fig. [fig surfaceBG bleaching]. This useful in that the background can be reduced by pre-bleaching of the window. However, the bleaching behavior also presents a nuisance. Frequent Xe-only deposits must be made in order to establish proper background subtraction, as historical bleaching and possible slight movements of the laser to differently bleached positions can cause variation in the background over en experiment period. Spacial variation in the background is especially cumbersome in a laser scanning experiment, as discussed in xxx. Given these behaviors, it is possible that the surface background is caused by a species which freezes to the window, or something which coats the window and fluoresces more at lower temperatures. In either case, the bleaching could be explained by evaporation with laser heating, or by optical pumping of the species into a metastable state.

### 4.3. IMAGING

Though the imaging spectrometer can produce spacial images with the 0-order grating reflection, better collection efficiency and imaging quality are achieved by removing the

spectrometer and imaging directly onto the CCD. Band-pass filters are used to pass the desired Ba fluorescence peak(s) while attenuating laser scatter and sapphire fluorescence.

An example image of the focused 570 nm dye laser passing through a c-plane sapphire window of thickness 0.5 mm, using the 620-nm band-pass filter on the fluorescence, is shown in Fig. [fig example image]. With  $4\times$  magnification, each pixel is  $5\text{ }\mu\text{m} \times 5\text{ }\mu\text{m}$ . The laser's path through the window is faintly visible by the tail of the broad  $\text{Cr}^{3+}$  fluorescence in the 620-nm region (see Fig. [fig band passes on spectrum with BG and Ba fluorescence shown] for regions passes by various filters). The laser is focused at the top surface of the window, which faces the ion beam. The surface background is seen on both surfaces. The observed size of the laser spot, with a  $1/e^2$  radius of about  $12\text{ }\mu\text{m}$ , is larger than the  $2.06\text{ }\mu\text{m} \times 2.66\text{ }\mu\text{m}$   $1/e^2$  laser spot size. Aberrations and vibrations in the collection optics could contribute to this inability to reach the diffraction limit in imaging.

**4.3.1. VIBRATIONS AND EFFECTIVE LASER REGION.** Relative vibrations between the laser and sapphire window will affect the number of Ba atoms exposed, increasing the effective laser spot size. This was studied by observing the position of a “dust spot” (a highly scattering feature on the sapphire window) relative to the position of the laser in an image on time scales down to 50 ms. An example of an image from this experiment is shown in Fig. [fig dust spot vibe image]. The 570-nm dye laser is somewhat de-focused, and the dust spot is illuminated by a de-focused **657 nm** diode laser. For each frame, 2D Gaussian functions are fit to locate the center of the laser spot and the dust spot in order to measure their relative position. The fit for the laser spot is restricted in y so that it is not affected by the bulk sapphire fluorescence path. The difference between dust spot and laser positions is plotted in Fig. [fig vibe vs. time with sine fit] for both x and y. Each exposure in this plot is 50 ms,

though readout time and camera shutter compensation time result in x s between frames. The best fit of a sine function results in a vibration frequency of x Hz. This is consistent with the audible frequency of the cryostat He pump.

To calculate an effective laser spot size due to this vibration, each difference (x,y) between laser and dust spot is used as the center of a 2D Gaussian, each with  $w_x = 2.06 \mu\text{m}$  and  $w_y = 2.66 \mu\text{m}$  to represent the laser spot. Such Gaussian functions are summed to all points to produce a distribution of summed laser exposure, shown in Fig. [fig vibe sum 2D]. The total area enclosed by a 1/e contour then represents the effective area. Since x and y movement is correlated, and since the movement is sinusoidal, the effective area is only about 2× the real laser spot size, and the effective laser region is **17.6  $\mu\text{m}^2$** .

The aforementioned is the most concerning vibration to understand. Vibration of the laser itself is included in that study. Vibration of the collection optics will affect the imaging resolution, but not the number of atoms being observed. These vibrations are minimized by stable mounts.

**4.3.2. IMAGING 577- AND 591-NM PEAKS.** First attempts at imaging small numbers of Ba atoms in a focused laser region were done with the 577- and 591-nm Ba fluorescence peaks together using a **579-nm** band-pass filter (Fig. [fig band passes on spectrum with BG and Ba fluorescence shown]). This filter has a 2" diameter, resulting in a factor of 4× more collection efficiency. Bleaching data is used to optimize laser intensity and exposure time to achieve maximal signal in frame 1, and to avoid hole bleaching. The fast bleaching of these peaks is the limiting factor on sensitivity. An image of  $\leq 10^4$  atoms ( $10^4 \text{ Ba}^+$  ions deposited into the effective laser region) is shown in Fig. [fig 590-lines image]. At only x total counts, this is near the limit of sensitivity. As discussed in xxx, detection of very small numbers of

atoms in these sites may require several infrared re-pump lasers. As a result of low total exposure, neither the sapphire nor the surface backgrounds are present in these images.

**4.3.3. IMAGING 619-NM PEAK.** At low laser intensities, the 619-nm peak appears minor compared to the 577- and 591-nm peaks. However, its much lower bleaching rate brings it to dominate over larger exposures, e.g. with the intensity of a focused laser at 0.01-0.1 mW over seconds or minutes. Experiments imaging small numbers of Ba atoms in SXe in a focused laser region with the 619-nm peak were successful down to the single-atom level.

An imaging experiment consisted of many different pulsed  $\text{Ba}^+$  deposits, each of which is evaporated after observation. This procedure is described in 3.3-3.5. A linear relationship between observed signal and ions deposited must be observed and repeated. Frequent Xe-only deposits are made, typically every three runs, to establish the local background, as the surface background depends on historical bleaching. The image of each deposit was analyzed by summing the counts of a 3-pixel×3-pixel ( $15\mu\text{m}\times15\mu\text{m}$ ) region centered on the laser spot in the SXe layer where it excites Ba atoms, i.e. at the top of the laser path in Fig. [fig example image of imaging intro]. A partial-bin integrator was implemented such that the 3-pixel×3-pixel region could be centered on the peak center of each run, which typically vary by about  $x \mu\text{m}$  between runs. Summed counts vs.  $\text{Ba}^+$  ions deposited into the laser region is shown in Fig. 4.8 for a combination of data sets from **two** separate days *maybe you want them separate (with each point having errors) by color to show agreement ... and try combining those other days. It will require scaling for intensity difference (spher. abb.) and time, and remember different  $I\times t$  might give different efficiency.* Each point has subtracted the averaged counts from the two surrounding Xe-only deposits. Some variation in signal is observed, likely due to spatial drifting of the ion beam, as this variation is larger on days

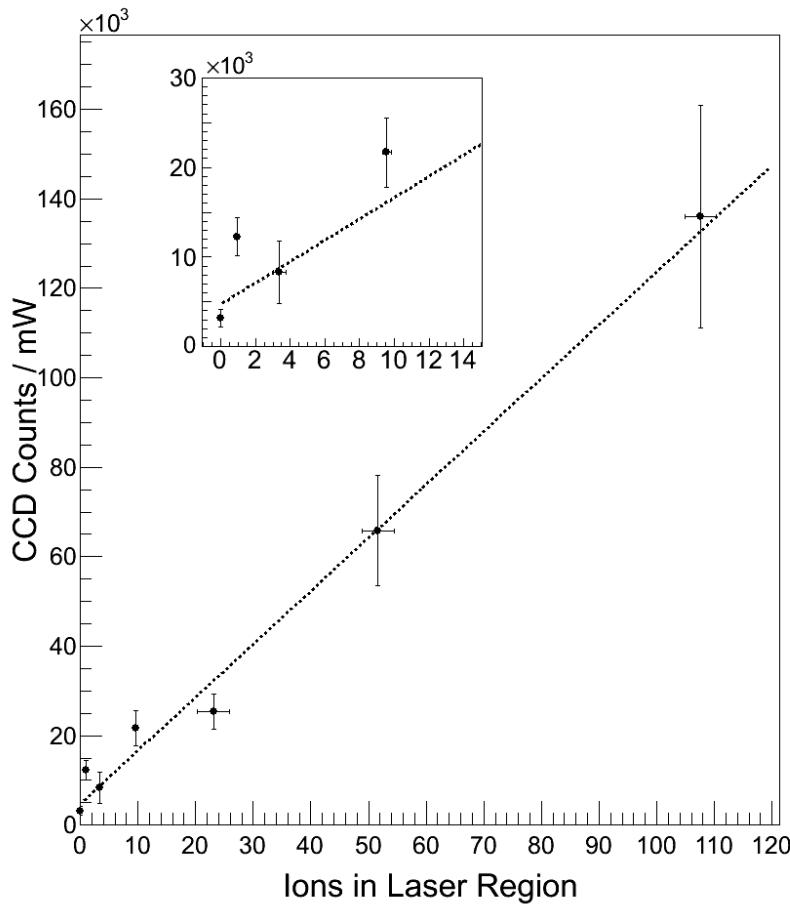


FIGURE 4.8. Better may be all points with stat errors instead of combined points, and try adding other days too: 5-26, and something before that (and after?). X errors should prolly just be systematic. Combined 2015-08-07 and 2015-09-16 with statistical errors. Move y-axis title over.

where larger beam drift is observed. Nonetheless, a clear linearity of observed signal vs. ions deposited, all the way down to single ions deposited. Recall that the number of ions is an upper limit on the number of atoms in the 619-nm peak matrix site. “0-pulse” deposits are also made, wherein Faraday cup 3 is retracted for the 1-s period, but the ion beam remains deflected with no pulses. This establishes the level of neutral Ba atoms making their way down the beam line from the source into the laser region. Signal at the few-atom level is observed from these deposits.

Variation in the background level, dominated by the surface background, is shown in Fig. [fig Xe vs run]. Variation is most likely caused by drift of the laser position on the window, to regions of different historical bleaching. Local variations are *1-atom level? less?* Positive signal after subtraction, even at the single-atom level, demonstrates that this variation is sufficiently low.

Xe-only images were also directly subtracted to produce images of 619-nm Ba fluorescence. In cases where the image has shifted slightly between runs, a binning redistribution was applied to do sub-pixel image shifting for better subtraction, though images which did not require this were preferred. 619-nm fluorescence images for several deposits of varying numbers of atoms are shown in Fig. [fig train].

It is important to note that no Ba signal is left behind after a sample is evaporated, even for large  $\text{Ba}^+$  deposits. Raw images (pedestal-subtracted) of  $\text{Ba}^+$  deposits and their preceding and succeeding Xe-only deposits are shown in Fig. [fig XeBaXe] for several runs. The lack of historical buildup of Ba is important for the implementation of this method of Ba tagging on a probe in nEXO.

**4.3.4. FURTHER CHECKS ON 619-NM PEAK.** The 619-nm peak, as well as the 670-nm peak, is attributed to neutral Ba by a few further tests. A large  $\text{Ba}^+$  deposit is compared to a deposit made with the Ba getter in Fig. [fig Ba+/Ba/Ar+]. The 619-nm and 670-nm peaks are observed with both sources, and since the getter produces only neutral Ba, they are attributed to neutralized ions in  $\text{Ba}^+$  deposits. Observation of the fluorescence with two different types of Ba sources is itself positive. In addition, observation of the fluorescence with the low deposit energy of the getter may bode well for the prospect of grabbing on a probe in LXe. Observation of a large deposit of  $\text{Ar}^+$  in SXe is also shown in Fig. []]. The lack

of fluorescence further disfavors a matrix-damage-related source of the fluorescence, such as color centers. To be thorough, imaging experiments of small Ar<sup>+</sup> deposits in SXe by pulsing were performed, otherwise identical to Ba<sup>+</sup> 619-nm imaging experiments. Summed counts from these deposits were consistent with background, as shown in Fig. [fig Ar+ pulse images].

*Whatever you did with the extra-filter-pass check (IR etc.) ... and was there something else?*

#### 4.4. CANDIDATE Ba<sup>+</sup> LINES/BLUE EXCITATION

Since the fraction of neutralization of Ba<sup>+</sup> deposited in SXe is not known to be 100%, in this system but particularly in future tests of grabbing out of LXe, detection of single Ba<sup>+</sup> in SXe is still of interest. Preliminary studies of fluorescence from Ba<sup>+</sup> deposits were performed by blue laser excitation with the C480 dye laser.

11 K vs 50 K (look back)? I think so... so, you might start with Shon's lines, and then say that, similar to with Ba, more was found with 50 K deps (right?), and then talk about excitspec, and then get into the 530/630 identical excitspec as well as bleaching, and so say they are the candidates. others are unidentified.

Mention 520 nm peak.

BaHx peaks – discovered/identified by Shon, but further attributed to H by temp and leak rate (BaSpec).

## CHAPTER 5

# CONCLUSIONS

### 5.1. FUTURE STEPS

Talk about whatever Chris will do next ... ions? Other sites? Populations?

Talk about Adam's work, show pics and stuff.

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