

Laves polyhedra in synthetic tennantite, $\text{Cu}_{12}\text{As}_4\text{S}_{13}$, and its lattice dynamics

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Abstract

Synthetic tennantite, $\text{Cu}_{12}\text{As}_4\text{S}_{13}$, is the analogue of an abundant mineral that belongs to the tennantite-tetrahedrite group with a low lattice thermal conductivity. Combined data from high-quality X-ray diffraction, electron microscopy (STEM-HAADF), Raman spectroscopy, as well as DFT calculations are used to analyze the peculiarities of structure stability and dynamics. Atomic displacement parameters and low-energy optical phonon modes are discussed within the context of the structure variations and peculiarities of the charge distributions. The latter indicates that the tennantite structure tends to conform to the covalent polar bonds and its charge distribution significantly affected by the atomic shifts in Laves polyhedra. According to the DFT calculations, there is a population of stable model structures with

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varying shifts of copper atoms in Laves polyhedra, with total energies being very close (within 0.24 eV), which can explain the observed behavior of the atomic displacement parameters. A specially designed technique used for the experimental analysis of the atomic displacement parameters revealed Einstein characteristic temperatures in the tennantite structure to be in the range of 50–190 K, attributed to low-energy optical phonon modes.

Highlights

- Synthetic tennantite $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ structure consists of disordered Laves polyhedra
- DFT calculations reveal a mixed valence of Cu atoms, and Laves polyhedra distortions
- Jahn-Teller distortions give rise to a set of unit cells with close ground energies
- Rattling vibration modes refer to Einstein temperatures of 74, 104, 115, 185K

Keywords: tennantite $\text{Cu}_{12}\text{As}_4\text{S}_{13}$, thermoelectric materials, DFT, anisotropy, optical spectroscopy, X-ray diffraction, computer simulations,

1. Introduction

The tennantite-tetrahedrite group represents a large class of inorganic compounds including complex copper sulfosalts with the general formula $\text{Cu}_{10+x}\text{M}_{2-x}\text{X}_4\text{S}_{13}$, where M is a transition metal (Zn, Fe, Hg, etc.), and X is a pnictogen (As, Sb, or Bi) [1], with the pure tennantite itself represented by

the formula $\text{Cu}_{12}\text{As}_4\text{S}_{13}$. Despite being known for over two centuries, these minerals are still under investigation, both due to structural peculiarities and potential applications. Arsenic- and antimony-containing compounds of this class are the advanced thermoelectric materials with a low lattice thermal conductivity[2, 3]. For example, the studies of tetrahedrite doped with Mn, Fe, Co, Ni, Zn, Te, Se showed ZT values in the range from 0.2 to 1.13[4, 5, 6, 7, 8, 9].

The tennantite structure treated as sphalerite-type and belongs to $I\bar{4}3m$ space group[10, 1]. There are three copper position: Cu1 is tetrahedrally coordinated by four sulphur atoms, Cu2 is coordinated by three sulphur atoms and Cu21 is tetrahedrally coordinated by three sulphur and arsenic atoms. The one arsenic is coordinated by three sulphur atoms. Cu21 and Cu2 atoms possess very large atomic displacements perpendicular to the plane of the triangle (Fig. 1 right). In a case of tennantite- and tetrahedrite-based materials, it is generally considered, that a low lattice thermal conductivity is caused by hybridized Cu $3d$ and chalcogen p orbitals[11]. Such a hybridization leads to a bonding asymmetry, and out-of-plane low frequency rattling modes, which are quasi-localized and anharmonic[12, 13].

Some authors note[14, 15], that asymmetric bonds give rise to magnetic properties and, in some cases, to antiferromagnetic (AFM) ordering[10, 16]. In both kinds of materials, deviations from the Curie-Weiss law in the temperature dependencies of the magnetization were revealed: at 124 K[15] in tennantite, and at 84 K[17] in tetrahedrite.

According to the structural study, these deviations take place due to the distortion of Cu_6S octahedra (Laves polyhedra)[15]. The distortion at 84

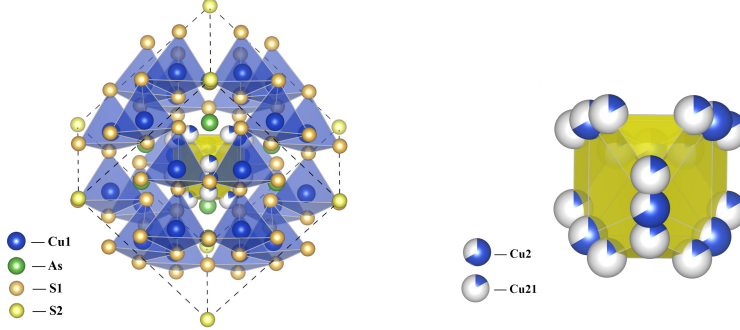


Figure 1: Left — the view of the $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ tennantite structure and of coordination polyhedra around the Cu1 (blue) and the Laves polyhedra (yellow); right — the Laves polyhedra with Cu2 and Cu21 atoms with occupancy.

K in tetrahedrite is reported to lead to either the symmetry lowering from $I\bar{4}3m$ to $I\bar{4}2m$ space group accompanied by the displacement of S atoms[18], or to a shift of Cu2 atoms towards S_3 triangle plane with S_2 displacement[17] in the same symmetry, and space group.

Recently, we performed a multi-temperature (90–298 K) structural study of a synthetic tennantite[10] and found that below ca. 120 K the copper atom distribution in tennantite changes from a more statistically delocalized Cu2 a well-defined split into Cu2 and Cu21 positions without a first-order phase transition. This effect is accompanied by As1 and S2 atoms displacement, which was also observed in ref. [18]. We also investigated the low- and room-temperature electronic structures of a synthetic tennantite based on the DFT calculations, which confirmed the second-order transition from the low-temperature AFM to the room-temperature PM state[10].

However, the relationships between the peculiarities of the tennantite

crystal and its electronic structures and thermoelectric properties, including the behavior of Laves polyhedra in the structure, were left aside in the previous paper[10]. In this paper, we focus on these problems, and investigate the tennantite cell energy distribution, and the charge density differences for different Cu positions in Laves polyhedra. Also, we consider low-energy phonon modes, and peculiarities of the atomic displacement parameters in the temperature range of 85–293 K, the same as in Ref. [19], in a context of the cell energy distribution. To clarify distortions in Laves polyhedra, a study of the spherical single-crystal sample was performed and the high-angle annular dark-field (HAADF) image was obtained.

2. Materials and methods

2.1. Synthesis

High quality single crystals of $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ for the XRD analysis were grown using a two-step procedure. At first, bulk $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ has been prepared by a high-temperature technique from copper (99.5 %), arsenic (99.5 %), and sulphur (99.5 %). Then, the ingots were subjected to directional recrystallization by the Bridgman-Stockbarger technique in a two-zone furnace. The temperature in the recrystallization zone was chosen 40 K above the tennantite melting point. A speed of the ingot motion was 0.5 mm/hour, with the temperature gradient in the recrystallization zone about 8–10 K/mm.

2.2. Single-crystal X-ray diffraction

A spherical shape single-crystal sample of $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ was prepared for the anisotropic extinction study[20]. Enraf-Nonius CAD-4 diffractometer

with AgK_α and graphite monochromator was used. The spherical shape single-crystal structural study results are presented in Appendix, they are compared with the structural data from the polyhedra defined form sample[10].

2.3. Analysis of temperature dynamics of atomic displacements

The temperature dynamics of an atomic displacement from the previous study[10] were used to estimate Einstein characteristic temperatures by the mixed Einstein-Debye model using DebyeFit program[19]. The data are based on the study of $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ single-crystal sample at 85, 115, 180, 250, and 293 K. The sample has a rectangular shape with the dimensions of $0.456 \times 0.31 \times 0.247 \text{ mm}^3$. Xcalibur diffractometer equipped with a CCD EOS S2 detector (Rigaku Oxford Diffraction), MoK_α and Cobra Plus (Oxford Cryosystems) with nitrogen gas flow were used. The original experimental methods and data processing are presented in Appendix in details.

2.4. Theoretical calculations

Calculations of the stability of possible tennantite polytypes were performed based on Density Function Theory (DFT)[21] within the generalized gradient approximation (GGA) using Perdew-Burke-Ernzerhof exchange-correlation functional[22]. We used the projector augmented wave method[23] with the periodic boundary conditions as implemented in Vienna Ab-initio Simulation Package (VASP)[24, 25, 26, 27]. The plane-wave energy cut-off was set to 450 eV. To calculate the equilibrium atomic structure, the Brillouin zone was sampled according to the MonkhorstPack scheme[28] with a $6 \times 6 \times 6$ grid in the k-space. The convergence criterion for the structural relaxation was set at 0.1 meV. Both the unit cell symmetry and parameters were allowed

to relax. To facilitate unconstrained structural optimization, all the starting structures were converted to $P1$ space group.

2.5. Electron microscopy

High-resolution electron microscopy images were taken using FEI Titan 80–300 transmission electron microscope in the HAADF–STEM mode (scanning translucent) with an acceleration voltage of 300 kV. The sample for microscopy was prepared using FEI Helios 600 focused ion beam setup. HAADF image was fitted by Atomap[29] package in Python.

2.6. Raman Spectroscopy

Raman spectroscopy study was performed with a spectrometer TRIAX-552, equipped with Peltier TE-cooled detector CCD SPEC 10 (Princeton Instruments). A grating of 600 lines mm^{-1} , and a $50\times$ lens Mitutoyo M Plan Apo SL50 (a numerical aperture of 0.42) were used. Raman spectra was excited with 514.5 nm argon-ion laser Spectra-Physics Stabilite 2017 with 1 mW output power on a sample. An open argon gas flow was used to reduce air modes. Calibration was performed using 520.5 cm^{-1} Raman peak of a polished silicon wafer. For observation of lines near the laser line we used 3 ONDAX 114-ER297-001 fiber Bragg gratings. Fittings of the Raman spectra were performed by the LMFIT[30] package in Python. The fitting model was based on 6 pseudo-Voigt functions and a polynomial function for the background.

Table 1: Anisotropic extinction parameters in $\text{Cu}_{12}\text{As}_4\text{S}_{13}$.

$g_{11} \times 10^4$	$g_{22} \times 10^4$	$g_{33} \times 10^4$	$g_{12} \times 10^4$	$g_{13} \times 10^4$	$g_{23} \times 10^4$
1.4(4)	4.2(8)	2.7(4)	1.0(3)	0.2(3)	-1.1(5)

3. Results and discussion

3.1. Remarks on the structure of synthetic tennantite

Study of the high-grade single crystal allowed us to re-investigate the crystal structure of $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ with a high precision. The structure was refined in a way similar to described early[10]. The sum of occupancies for Cu2, Cu21 and its symmetry related site were constrained to 1. Due to rather large distance between these atoms their ADP were refined independently. The maximum entropy method (MEM) refinement gave the result identical to [10]. Our data confirms that the compound contains 12 copper atoms per formula unit, and shows no evidences of a copper deficiency. Neither XRD, nor TEM data have shown detectable packing defects.

Due to the crystal growth method used, anisotropic extinction in the sample takes place. Extinction parameters are presented in the Table 1. The disorder of Cu21 is observed in the analyzed HAADF image of the sample in [110] zone (Fig. 2). The electron diffraction, and overlay of HAADF image with the tennantite crystal [110] zone generated by VESTA[31], are shown in Appendix. The processed HAADF image (Fig. 2 right) shows the ellipticity for the Cu/As atomic columns, which is a measure of the atomic column elongation[29]. The atomic column elongation is explained by the blurring of the intensity distribution, and a large atomic displacement. A similar inhomogeneous electron density was observed in tetrahedrite[32], and

tetrahedral framework structure[33]. Also, the anisotropy may be caused by the crystal growth conditions or/and a crystal cell distortion. If so, then the structure would have a diagonal distortion or/and have more or less than 12 copper atoms per cell, which is not the case.

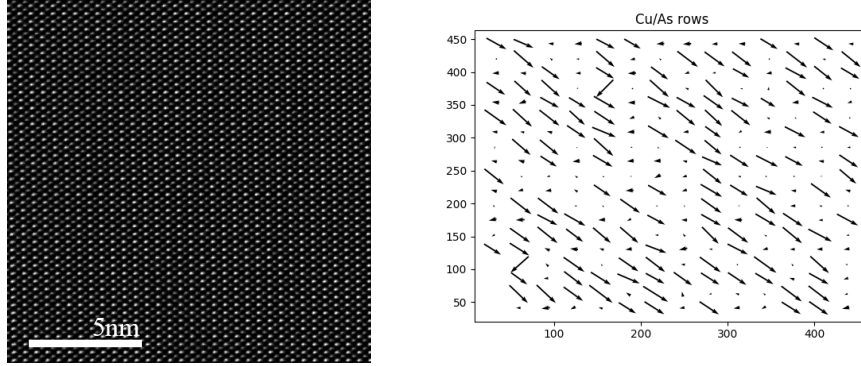


Figure 2: Left — the HAADF image of tennantite crystal viewed from the $[110]$ zone; right — the processed HAADF image with the ellipticity for the Cu/As atomic columns. The analysis implied quantifying the positions and shapes of atomic columns based on Atomap[29].

3.2. Atomic displacement parameters analysis of the temperature dynamics based on diffraction data

The atomic displacement parameters (ADPs) were analyzed by DebyeFit program using the structural data. The DebyeFit estimates the Debye or/and Einstein characteristic temperature in crystals from the equivalent atomic displacement parameters obtained at different temperatures.

At fitting, ADPs are considered as a combination of quantum zero vibrations, thermal vibrations, and static shifts[34, 35, 36]. The thermal expansion of the crystal, and the corresponding change of the atomic vibra-

tion frequencies depend on the temperature, and may be described using a quasi-harmonic approximation[37, 35]. As a result ADP, is described by $u_{\text{calc}} = xu_E + (1-x)u_D + u_{\text{static}}$, where xu_E and $(1-x)u_D$ are Einstein and Debye parts, u_{static} is a static shift of an atom from its site in the structure model[19]. The result of tennantite ADPs fitting is presented in the Fig. 3, and Table 2. The average oscillation frequency of the selected atom, and its characteristic Einstein temperature, T_E , can be obtained assuming $x = 1$. The maximal oscillation frequency of the selected atom and its characteristic Debye temperature, T_D , corresponds to the case $x = 0$. Distributions of zero-point oscillations, u_{0E} or u_{0D} , were obtained applying $T = 0$ in the Einstein or Debye formulas. Thus, the contribution of Einstein or Debye parts to atomic displacements consists of zero quantum vibrations, u_{0E} or u_{0D} , and a temperature-dependent part, $u_E(T)$ or $u_D(T)$, which can be associated with the thermal motion of atoms. That is, $u_E = u_{0E} + u_E(T)$, and $u_D = u_{0D} + u_D(T)$. The shift of an atom from its equilibrium position due to any reason other than thermal vibrations is the sum of zero vibrations and static shift: $u_{\text{shift}} = u_{0E} + u_{\text{static}}$ or $u_{\text{shift}} = u_{0D} + u_{\text{static}}$ [19].

A good agreement between model ADP, u_{calc} , and equivalent experimental ADP, u_{eq} , can be obtained only if (1) the temperature of the crystal is correctly determined during the measurements; (2) the static shift of the atoms is taken into account, i.e. when using the extended model with inclusion u_{static} term. The fitting results are presented in Figure 3.

Large static values for Cu2 atom (Table 2) indicate the potential instability of this position. According to structurel study of tennantie Cu2[18] position may be describied as 6 differnt Cu position. But according our

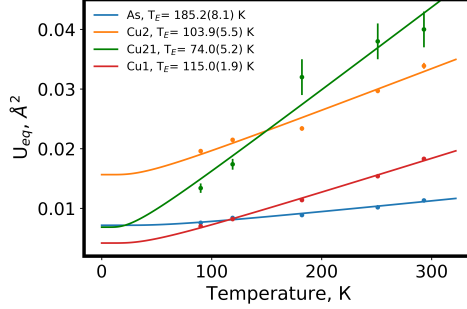


Figure 3: Equivalent parameters U_{eq} of the mean-square displacements of atoms in tennantite. Points are the experimental data. The solid curves show the fit of the temperature dependences according to the expanded Einstein model.

Table 2: Einstein temperatures, dispersions of zero-point oscillations u_{E_0} , u_{static} , and u_{shift} for atoms in tennantite. $R = \sum |u_{eq} - u_{calc}| / \sum u_{eq}$ is the residual factor of the LS procedure.

	T(Einstein) K, cm^{-1}	$u_E \times 10^4, \text{\AA}^2$	$u_{static} \times 10^4, \text{\AA}^2$	$u_{shift} \times 10^4, \text{\AA}^2$	R, %
Cu ₁	115.0(1.9), 79.9	3.3	0.9(0.4)	4.2	1.89
Cu ₂	103.9(5.5), 72.2	3.7	12.0(1.5)	15.7	3.28
Cu ₂₁	74.0(5.2), 51.4	5.2	1.7(4.0)	6.8	8.06
As ₁	185.2(8.1), 128.7	1.7	5.4(0.3)	7.2	2.20

study even after dividing Cu2 position into Cu2 and Cu 21, the Cu2 position remains highly disordered and there are no reasons for adding additional positions. This disordering is reliably described by the anharmonic model of atomic displacements using the Gram-Charlier expansion to the 4th order.

3.3. Analysis of anharmonic vibrations of copper atoms through the ADP and Raman spectroscopy

The Raman spectrum of the crystalline tennantite consists of the following well-defined peaks: ν_1 , ν_2 , l_1 , l_2 at 374.5, 340, 64, and 122 cm^{-1} respectively,

and a broad peak ν_4 at 320 cm^{-1} . Peaks between 200, and 400 cm^{-1} refer to (Sb, As) S_3 groups modes[38]. Position of ν_3 was found by the spectra fitting. The fit result is presented in Figure 4, and in Table 3. Peaks below 200 cm^{-1} are related to the lattice dynamics[39].

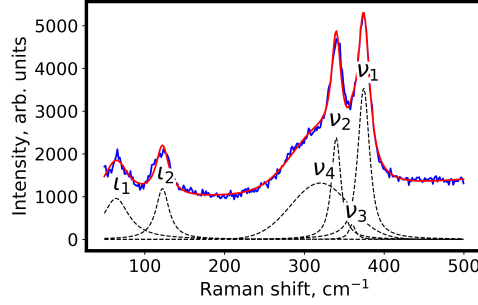


Figure 4: The Raman spectrum of the crystalline tennantite. Peaks between 200, and 400 cm^{-1} refer to (Sb, As) S_3 groups modes[38]. Peaks below 200 cm^{-1} are related to the lattice dynamics[39]. Blue — experimental data, red — the superposition of modes, dotted — fitted peaks

Within **approach above**, the Einstein solid model is used to describe independent harmonic oscillators. If so, the low energy spectrum should consist of peaks with the Einstein's energies, which we found by DebyeFit (Table 2). Results of calculations are presented in Figure 5. The Einstein's energies from structural study were used as the initial points for the fit, and in Table 3 the initial and fitted wavenumbers are presented. The low energy l_1 and l_2 peaks are well described by Einstein's energies of Cu2, Cu21, Cu1, and As (66, 40, 79, and 122 cm^{-1} respectively). **Same modes were observed and analyzed in tetrahedrites[12, 11] with energies approximately 4, 9 and 18 meV (33, 72, 145 cm^{-1}). The energies of tetragidrite linked to the trigonally-**

Table 3: Initial and fitted Einstein's energies. Initial energies are Einstein's energies from structural study.

	Cu2	Cu21	Cu1	As
Initial Einstein's energies, cm^{-1}	72	51	80	129
Fitted Einstein's energies, cm^{-1}	66	40	79	122

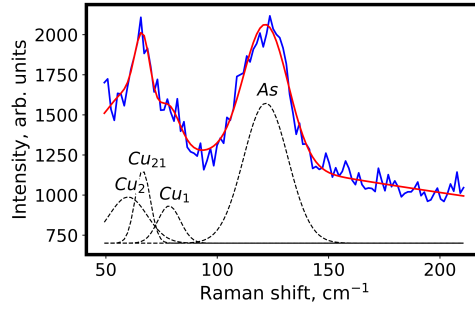


Figure 5: The Raman spectrum of the tennantite crystal below 200 cm^{-1} . Experimental peaks are considered as the peaks with Einstein's energies. Blue — experimental data, red — modes based on the Einstein's energies fit, dotted — fitted peaks.

bonded Cu atoms: the lowest energy is linked to Cu2 (they do not divide to Cu2 and Cu21), the middle — to Cu1 and the highest — to Sb. That in good agreement with our results. Thus, the reasons for the strong anharmonic behavior, described in the papers, is a dynamic disorder of the copper atoms and arsenic atoms. Consequently, this leads to out-of-plane low frequency rattling modes, and a low lattice thermal conductivity.

3.4. Theoretical analysis of possible polytypes and morphology of Laves *polyhedra*

Using the DFT calculations, we tested an assumption that the synthetic tennantite crystal structure is not the most energetically stable, and could feature various polytypes. As an initial point, we considered tennantite structure described in $I\bar{4}2m$ space group (henceforth called *reference structure*). The unit cell contains Laves *polyhedra* formed by copper atoms around sulphur ones (Figure 6, number 0). By shifting copper atoms by ca. 1 Å, we produced a population of structures with varying shapes of Laves *polyhedra*, which were then subjected to the procedure of unconstrained structural optimization. In the DFT calculations, where we cannot have Cu2 and Cu21 at the same time since we cannot have part of an atom occupying a position within a single polyhedron, we prefer to only use Cu2 notation, but distinguish their positions before and after optimization. The Laves *polyhedra* morphologies before, and after the optimization, are shown in Figure 6. The total energy differences between the initial tennantite and optimized structures are shown in Figure 7.

Our calculations show, that the experimentally determined synthetic tennantite structure is not the most energy favorable one at 0 K.

We observe 8 structures in our population with the energies less than the starting one. They correspond to the structures No's 2, 3, 9, 11, 14, 17, 18, and 19, with the last one possessing the the lowest energy. The coordination of the central sulphur atom deviates quite a lot from an ideal octahedral, with the energy gain being achieved by shifting one (structure 9), two (structures 17, 18), or three (structures 2, 3, 11, 14, 19) atoms.

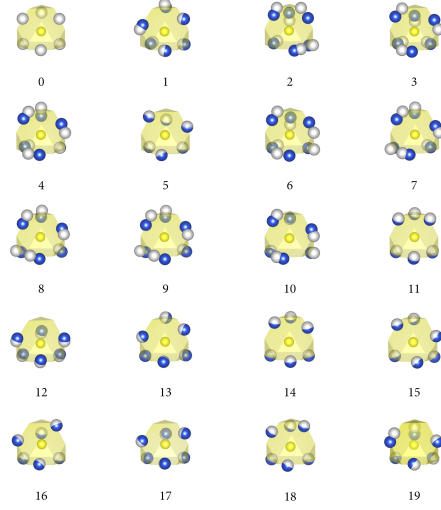


Figure 6: Geometries of SCu_6 Laves **polyhedra** before, and after structural optimization. Sulphur atoms are shown in yellow, copper atoms before optimization are shown in grey, and after optimization in blue. VESTA[31] was used for visualisation.

Table 4: The difference between the reference structure energy, and the unit cell energy of tennantite with different Laves **polyhedra**.

Structure, number	0	1	2	3	4	5	6	7	8	9
The difference of unit cells energies, eV	0.00	0.23	-0.02	-0.02	0.05	0.13	-0.00	0.23	0.06	-0.01
Structure, number	10	11	12	13	14	15	16	17	18	19
The difference of unit cells energies, eV	0.16	-0.02	0.05	0.03	-0.02	0.02	0.12	-0.02	-0.03	-0.04

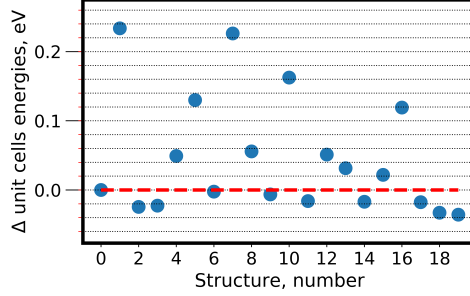


Figure 7: The difference between the reference structure energy and the unit cell energy of tennantite with different Laves **polyhedra**. Horizontal red line shows the reference unit cell energy.

The analysis of atomic charge distribution shows that the optimized structures can be divided into three groups: 1) with As charge of ca. +3 (reference compound also falls into this category), 2) with As charge less than +1, and 3) with intermediate As charge (between ca. +1 and +2). Charge density on arsenic affects substantially the respective charge density at 1st sulphur position (see Appendix), and has a minor effect on Cu2 charges, while Cu1 is virtually unaffected by any change. The other sulphur atom shows minor charge fluctuations, and they appear to be not directly associated with the state of arsenic.

Each of these groups has structures with an energy gain over the reference structure, and each of them has a certain split of a positive charge on Cu2 atoms, while Cu1 atomic charges remain constant throughout. Thus, we can say that the splitting of the Cu2 position into Cu2 and Cu21, as evident from the XRD structure analysis, facilitates the differences in charge distribution. Hence, the energy gain from forming Cu21 position leads to slightly differ-

ent electronic states of Cu2 and Cu21 atoms, which, in turn, leads to the formation of asymmetric bonds, as was pointed out for tetrahedrites[11, 40]. We can also observe certain similarities between tennantite and inorganic compounds featuring atoms with $3d^9$ configuration, and having asymmetric Fermi surface. The resulting complex charge distribution, and variances in the copper atom states, may, in turn, be the reason for complex exchange interactions and the reported AFM ordering[10] in tennantite. Moreover, the same structure features have been described for tetrahedrite[41]. The authors noted, that understanding these features provides a key to descriptions of the phonon-limited electrical resistivity, and the Lorenz number.

It must be noted that the majority of the favorable structures belongs to group 2, i.e. more covalent structures with both arsenic and sulphur atoms showing the relatively low positive and negative charges, respectively. Interestingly, the reference structure is significantly more ionic in this respect. Also, we must point out that, according to the DFT calculations, several structures have energy minimum within hundredths of eV (Table 4), i.e. only several kJ/mol ($1 \text{ eV} \sim 96.5 \text{ kJ/mol}$), which is on the limit of the accuracy of the method. This indicates, that the tennantite experimental structure may be an averaged result of several variations of the copper atoms shifts.

4. Conclusions

In conclusion, the structural variety of synthetic tennantite, $\text{Cu}_{12}\text{As}_4\text{S}_{13}$, leads to a dynamic disorder of the copper atoms, asymmetric bonds, and different charge distribution. This brings to the out-of-plane low frequency rattling modes, well described by Einstein characteristic temperatures: 74,

104, 115, and 185 K. The peculiarities of the tennantite crystal and electronic structures lead to asymmetric Fermi surface, thus making tennantite–tetrahedrite group compounds attractive research objects.

Our extensive structural studies revealed a strong electron density anisotropy of the bulk sample, with no evidence of a distortion or local defects.

The specially-designed technique was tested for the experimental ADPs analysis, and we have demonstrated a new method of the experimental study of a crystal structure.

5. Acknowledgments

This research was partially supported by Russian Science Foundation (Grant No. 19-13-00451). The calculations were carried out using the equipment of the shared research facilities of the HPC computing resources at Lomonosov Moscow State University. X-ray diffraction study was performed using the equipment of the Shared Research Center FSRC Crystallography and Photonics RAS and was supported by the Ministry of Science and Higher Education of the Russian Federation (project RFMEFI62119X0035). X-ray diffraction part of this work was supported by the Ministry of Science and Higher Education of the Russian Federation within the State assignment of the Federal Scientific Research Center Crystallography and Photonics of the Russian Academy of Sciences. Electron microscopy was carried out using the equipment of the Resource Centre Nanosond National Research Center Kurchatov Institute. We thank I. A. Perezhogin for valuable comments and Professor A. N. Babushkin for the tennantite ingot.

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